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A review of graphene derivative enhancers for perovskite solar cells

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Due to the finite nature, health and environmental hazards currently associated with the use of fossil energy resources, there is a global drive to hasten the development and deployment of renewable energy technologies. One such area encompasses perovskite solar cells (PSCs) that have shown photoconversion efficiencies (PCE) comparable to silicon-based photovoltaics, but their commercialisation has been set back by short-term stability and toxicity issues, among others. A tremendous potential to overcome these drawbacks is presented by the emerging applications of graphene derivative-based materials in PSCs as substitutes or components, composites with other functional materials, and enhancers of charge transport, blocking action, exciton dissociation, substrate coverage, sensitisation and stabilisation. This review aims to illustrate how these highly capable carbon-based materials can advance PSCs by critically outlining and discussing their current applications and strategically identifying prospective research avenues. The reviewed works show that graphene derivatives have great potential in boosting the performance and stability of PSCs through morphological modifications and compositional engineering. This can drive the sustainability and commercial viability aspects of PSCs.

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1. Introduction

As the global village advances in machinery and technology that sustains and enables humankind to live comfortably, environmental and health issues are also rising from the extensive dependency on highly polluting fossil-based energy resources. Fossil resources currently provide 81% of the global total energy supply.¹ Besides their contribution to climate change, the combustion of fossil-based energy resources is also causing hazardous effects on the health of the world's population. For example, 18% of the total global deaths in 2018 were from pollution-related ailments emanating from the use of fossil-based energy resources.² According to 2018 statistics from the World Health Organization (WHO), the use of fossil-based energy resources has mostly affected children (>40% environmental-related diseases and >88% of the health complications from climate change were experienced by children under five years).³ Economic and health costs (work absenteeism, lives lost and premature deaths) due to air pollution from burning fossil fuel energy resources ascended to \$2.9 trillion in 2018.⁴ Since fossil-based energy resources are

non-renewable and causing alarming damaging outcomes, other alternatives need to be developed to dependable levels.

One hugely important renewable energy technology is photovoltaics, with a global energy output growth of 60% between 2000 and 2016.⁵ For instance, in Europe, where 70% of the global total installed solar cells are located, annual PV installation rose from 58 MW to 222 GW between 2000 and 2015.⁶ In this ambit, perovskite solar cells (PSCs) in particular, are not only a rising energy technology but a cheaper option as well. In 2016 the hypothetical levelized cost of electricity (LCOE) for a perovskite solar cell (PSC) with PCE of 20% and a projected PSC life span > 15 years was calculated to be 3.5 US cents per kW h, whilst the LCOE for traditional energy resources, such as oil and gas, was in the range US 7.04–11.90 cents per kW h.^{7,8} PSCs have emerged with great potential to compliment market-leading Si PV due to their high PCE and potentially low manufacturing costs.^{8,9} Additionally, in 2017, Song *et al.*¹⁰ stated that the production cost of a standard PSC was US \$31.70 m⁻² (\$6.80 m⁻² was PSC processing costs and \$24.90 m⁻² was the balance of module (BOM) components). This means the manufacturing cost of PSCs, after excluding BOM component expenditure, is lower than that of other thin-film solar cells (CIGS: US \$29 m⁻² and CdTe: US \$27 m⁻²).^{8,10} This analysis also infers that the major contributor to PSC costs is the BOM components.

Additives and substitution of BOM components with lower-cost alternatives is a promising approach in balancing the

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possible trade-off between cost and PCE. In a PSC, 76% of the total expenditure is from materials; however, the cost of the material in the perovskite unit cell structure and active material contributes only 7.6% to this.^{10,11} Examples of expensive BOM components in PSCs include noble metals, such as Au, conductive substrates, and hole transporting layers (HTLs), such as spiro-OMeTAD.^{12,13} The use of lower-cost, alternative sustainable materials or strategies (such as remanufacturing) will be crucial in minimising LCOE and improving the chance of commercial success.

Although solar cells have the potential to meet clean global energy demands, the waste from photovoltaics after reaching their end of life is rising to alarming levels. For example, the cumulative photovoltaic waste is projected to reach 78 million tonnes by 2050.⁶ One strategy to help alleviate this problem is the greater use of eco-friendly materials that can reduce both bio-incompatible waste accumulation and recycling costs. Coupled with this waste problem is the growing use of primary materials, with renewable energy technologies requiring significant amounts of critical raw materials. To ensure rapid deployment of renewable technologies is not disrupted, materials with inherent natural abundance are needed. Graphene derivatives are one such class of materials as they can be obtained from the abundant and naturally occurring graphite and biomass. To date, sustainable additives such as graphene derivatives have principally shown potential in aiding exciton dissociation, conductivity and long-term stability in PSCs (reviewed articles in Table 1).

This review focuses on the application and suitability of graphene-based derivative additives as components of the substrate, active-, charge transport- and blocking-layers in PSCs. We pay particular attention to their influence on the stability and performance of PSCs, and reflect on any drawbacks these graphene derivative materials may pose when incorporated in PSCs.

2. Perovskite solar cells

Perovskites have the general formula ABX_3 , where A is the larger cation, typically CH_3NH_3^+ (MA^+), $\text{HC}(\text{NH}_2)_2^+$ (FA^+), Cs^+ or mixed cations, such as FAMA, which occupy the cubo-octahedral site shared with twelve halide anions X^- . B is the smaller cation (Pb^{2+} , Sn^{2+} , etc.) which is stabilised by the octahedral setup and shared with six halide anions, X^- .¹⁴⁻²⁰ An organic-inorganic hybrid perovskite is usually used in solar cells, and the methylammonium lead iodide (MAPbI_3) is the most common (Table 1). Perovskites have a direct optical band gap of typically 1.5 eV, although this can be tuned through the visible region by altering their chemical compositions.²¹⁻²³ The impressive performance of perovskites in solar cells is generally due to the high absorption coefficient ($>10^4 \text{ cm}^{-1}$), long charge-carrier lifetime and long charge-diffusion path ($>1 \mu\text{m}$).²⁴⁻²⁹ A further advantage is the compatibility with low-cost solution processing techniques. In addition to quality (morphology and uniformity), the film thickness of the perovskite layer is an important factor for maximising the efficiency of devices. The high absorption coefficient of perovskites results in thin-films of *ca.* 400 nm being suitable for

efficient photon capture. This is advantageous because it results in less use of materials; even though perovskites are typically composed of earth-abundant materials, this still benefits cost and helps reduce materials wastage.

The working mechanism of a PSC involves the absorption of light by the perovskite layer; upon excitation, excitons quickly dissociate at room temperature due to a very low exciton binding energy, resulting in free charges. Electrons are extracted at the electron transport layer (ETL), while holes are extracted at the hole transport layer (HTL) prior to transportation to the anode and cathode, respectively (Fig. 1).²¹ Interfacial electron transfer from the perovskite conduction band to the HTL and/or ETL surface states, together with interfacial electron transfer from the conduction band of the ETL to the HTL and/or perovskite, are possible recombination mechanisms in PSCs (Fig. 1).³⁰ The current review will, in later sections, discuss the potential of graphene derivatives to retard recombination in PSCs.

The PCE of PSCs has dramatically increased from about 10% in 2012³¹ to a certified value of 25.5% by 2019,^{32,33} which is comparable to the 26% PCE of monocrystalline Si-based solar cells.³³⁻⁴³ PSCs are well-positioned to be successfully commercialised since their PCE has surpassed other thin film-based solar cells (CIGS: 23% and CdTe: 22%).^{33,44,45} This significant improvement of PCE within the last decade is mainly attributed to morphological tailoring and compositional engineering of the PSC device layers.⁴⁶ Relative to the first two photovoltaic generations, PSCs have the potential to harness light at lower costs and are also associated with facile fabrication procedures.⁴⁷

Over 12 000 articles on PSCs had been reported by 2019, and since then, companies have been focusing on the commercialisation of the technology.^{19,40} PSCs have a great potential in finding a niche in the world energy markets because they are highly efficient, printable, can be noble metal- and HTL-free.⁴⁸ For example, PSCs have high potential in wearable devices due to the associated facile fabrication, high PCE and low costs.^{39,49,50} To date, the technology has already advanced to mini- and standard-sized modules.¹⁹ For instance, Yeo *et al.*⁵¹ reported PCEs in the 8.1–10% range for scalable flexible and rigid modules with an area of 10 cm^2 . However, PSCs are still not found substantially in the global markets because the technology is in the laboratory/industrial scale transition phase.^{21,48,52} Roll-to-roll technology is more practical for optimum substrate coverage at large-scale due to low fabrication costs, sample nature, device structure and manufacturing steps involved. The transformation from laboratory to large-scale roll-to-roll deposition still needs more insights on several aspects, such as solvent selection, annealing temperature, and film thickness control and deposition rate, to avoid a substantial decrease in PCE.^{53,54}

Additionally, the commercialisation and large-scale production of PSCs is also hindered by toxicity issues due to the use of solvents,⁴³ such as DMF,⁴³ γ -butyrolactone (GBL),⁵⁵ chlorobenzene and toluene,^{12,19} and Ag and Pb as counter electrodes.^{21,42} Notwithstanding the fact that the toxicity of $\text{Pb}(\text{II})$ salts is a problem, studies are yet to provide a lucrative prospective



Table 1 Illustrative applications of graphene derivatives in perovskite solar cells^a

Device structure	Conditions ^a	Stability		PCE decline (%)	J_{sc} (mA cm ⁻²)	V_{oc} (V)	FF (%)	Champion		Ref.						
		Time (d)	n/d					PCE (%)	FF (%)							
Spin-coating deposition of graphene-based materials																
As HTL component and electron blocking layer in p-i-n configuration																
Glass/ITO/GO/PEDOT:PSS/MAPbI ₃ /PCBM/rhodamine 101/LiF/Au	n/d	n/d	n/d	18.20	0.97	80.00	14.10	82								
Glass/ITO/GO/PEDOT:PSS/MAPbI ₃ /PCBM/carbon tape	Ambient	4	0	13.80	0.80	48.00	5.20	193								
Glass/Ag nanowire–GO/PEDOT:PSS/MAPbI ₃ /PCBM/2,9-dimethyl-4,7-diphenyl-1,10-phenanthroline (BCP)	n/d	1	0	15.43	0.87	70.90	9.62	122								
Glass/ITO/PEDOT:PSS–GO/(FAPbI ₃) _{0.85} (MAPbBr ₃) _{0.15} /PCBM/BCP/Ag	n/d	n/d	n/d	20.01	0.90	79.00	14.20	108								
Glass/ITO/PEDOT:PSS–GO/MAPbI ₃ /PC ₇₀ BM/Al	n/d	n/d	n/d	17.92	1.03	71.00	12.76	162								
Glass/ITO/PEDOT:PSS–NH ₃ –GO/MAPbI ₃ /PCBM/Bphen solution/Ag	Air	4	28	22.06	1.03	71.00	16.11	85								
Glass/ITO/PEDOT:PSS–PANI–GO/MAPbI ₃ /PCBM/rhodamine 101/Ag	Ambient, RH: 20%	80	70	22.89	1.05	75.40	18.12	94								
Glass/FTO/PANI–GO/MAPbI ₃ /PCBM/Ag	n/d	n/d	n/d	21.23	0.52	67.00	9.24	160								
Glass/ITO/PEDOT:PSS–RGO/MAPbI ₃ /PCBM/BCP/Ag	n/d	n/d	n/d	16.75	0.87	75.00	10.70	194								
Glass/ITO/PEDOT:PSS–RGO/MAPbI ₃ /PCBM/Al	n/d	n/d	n/d	17.10	0.95	65.00	10.60	195								
Glass/ITO/PEDOT:PSS–sulfur–RGO/MAPbI ₃ /PCBM/Ag	n/d	n/d	n/d	19.40	1.01	67.00	13.00	196								
Glass/ITO/PEDOT:PSS–sulfonic acid–RGO/MAPbI ₃ /PCBM/BCP/Ag	Ambient air	30	12	19.39	1.04	80.48	16.01	117								
Glass/ITO/PEDOT:PSS/poly(ethylene oxide)–GO/MAPbI _{2.5} Br _{0.5} /PCBM–MoS ₂ /Ag	AM: 1.5 G	17	6	22.83	1.14	73.80	19.14	197								
Glass/FTO/NiO/GO/MAPbI _{3-x} Cl _x /GO–Li/TiO _x /Al	Ambient air, RH: 20–38%	15	30	18.60	0.97	62.00	11.20	168								
Glass/ITO/PEDOT:PSS/fluorinated RGO/MAPbI ₃ /PCBM/BCP/Ag	n/d	n/d	n/d	19.10	1.01	76.2	14.70	51								
Glass/ITO/PEDOT:PSS/MAPbI ₃ /PCBM–RGO/poly[[(9,9-bis(3'-(<i>N,N</i> -dimethylamino)propyl)-2,7-fluorene)- <i>alt</i> -2,7-(9,9-diptylfluorene)] (PFN)/Ag	Continuous light, RH: >50	5	45	22.92	0.85	65.80	12.82	198								
Solely as HTL in p-i-n configuration																
Glass/ITO/GO/MAPbI _{3-x} Cl _x /PCBM/ZnO/Al	n/d	n/d	n/d	17.46	1.00	71.00	12.40	25								
Glass/ITO/GO/FA _{0.2} MA _{0.8} Pb(I _{0.8} Br _{0.2}) ₃ /PCBM/ZnO/Ag	RH: 65–75	4	60	21.00	1.00	71.00	14.90	118								
Glass/ITO/GO/C quantum dots/MAPbI ₃ /PCBM/BCP/Ag	Temperature (T): 25 °C RH: 25–30%	2	10	18.70	0.95	80.10	16.20	199								
Glass/FTO/GO/MAPbI _x Cl _{3-x} /TiO ₂ –Li–GO/Al	n/d	n/d	n/d	15.60	0.91	72.00	10.20	115								
Glass/FTO/N–GO nanoribbons/MAPbI ₃ /ZnO/Al	T: 20 °C, RH: 47%	2	11	17.42	1.00	71.30	12.41	200								
Glass/FTO/NH ₃ –GO/MAPbI _{3-x} Cl _x /PCBM/bathocuproine (BCP)/Ag	n/d	30	10	18.40	1.00	76.80	14.14	158								
Glass/ITO/RGO/MAPbI ₃ /PCBM/Ag	T: 25 °C, RH: 30%, no encapsulation	41	50	22.1	0.96	77.00	16.40	201								
Glass/ITO/RGO/Cu(i) thiocyanate (CuSCN)/MAPbI ₃ /PCBM/BCP/Ag	Bending cycles (cycles): 150	n/d	n/d	30.00	n/d	n/d	n/d		202							
Glass/ITO/MoO ₃ –RGO/MAPbI ₃ /PCBM/PCP/Ag	Continuous AM = 1.5 sun	4	10	18.21	1.03	76.10	14.28		202							
Glass/ITO/poly-(<i>N</i> -vinyl pyrrolidone)–RGO/MAPbI ₃ /PCBM	Encapsulated, RH: 30%	30	35	21.18	1.12	77.00	18.15	177								
	n/d	42	50	14.86	0.97	79.95	11.36	203								
In composite with ETL and as a hole blocking layer in n-i-p configuration																
Glass/ITO/ZnO–GO/MAPbI ₃ /Au	n/d	n/d	n/d	21.51	0.67	54.00	4.52	204								
Glass/ITO/compact TiO ₂ (c-TiO ₂)/m-TiO ₂ /Li–GO/MAPbI ₃ /spiro-OMeTAD/Au	1 sun	2.5	17	19.61	0.86	70.30	11.14	120								
Glass/ITO/SnO ₂ –N–GO/MAPbBr ₃ /spiro-OMeTAD/Au	Ambient air, T: room, RH: 25	n/d	12	18.87	1.17	74.93	16.50	35								
Glass/FTO/c-TiO ₂ /m-TiO ₂ –GO/mesoporous-ZrO ₂ /MAPbI ₃ /carbon	n/d	n/d	n/d	22.84	0.98	61.72	13.60	205								



Table 1 (Contd.)

Device structure	Conditions ^a	Stability		PCE decline (%)	J_{sc} (mA cm ⁻²)	V_{oc} (V)	FF (%)	Champion PCE (%)	Ref.
		Time (d)	n/d						
Glass/FTO/c-TiO ₂ /m-TiO ₂ –RGO/MAPbI _{3-x} Cl _x /Cu–Bu–phthalocyanine	Ambient air, T: room, RH: 28–32%	n/d	34	21.00	1.07	71.00	15.90	27	
Glass/FTO/c-TiO ₂ /m-TiO ₂ –RGO/MAPbI ₃ /spiro-OMeTAD–Li	n/d	n/d	n/d	22.00	0.93	70.70	14.50	26	
Glass/FTO/c-TiO ₂ /m-TiO ₂ –PANI–RGO/CsPbI ₃ –PbI ₂ /spiro-OMeTAD/Au	Encapsulated, T: 20–30 °C, RH: 20%	77.9	18	26.96	0.96	63.60	16.48	206	
Glass/FTO/Bl–TiO ₂ /rGO _{4-x} TiO ₂ /(FAPbI ₃) _{0.85} (MAPbBr ₃) _{0.15} /spiro-OMeTAD/Au	n/d	n/d	n/d	22.16	1.07	75.40	17.66	207	
As a constituent of the active layer in n-i-p configuration									
Glass/ITO/SnO ₂ /MAPbI ₃ –GO/spiro-OMeTAD/Au	n/d	n/d	n/d	23.73	1.07	69.14	17.59	208	
Glass/FTO/SnO _x /Cs _{0.05} (FA _{0.85} MA _{0.15}) _{0.95} Pb(I _{0.85} Br _{0.15}) ₃ –dodecylamine–GO/spiro-OMeTAD/Au	Room temperature, no encapsulation	40	30	22.10	1.10	81.00	21.10	106	
Glass/FTO/TiO ₂ /MAPbI ₃ –CNT–P3HT–GO/spiro-OMeTAD/MoO ₃ /Au	n/d	n/d	n/d	22.73	0.96	75.00	16.36	93	
Glass/FTO/c-TiO ₂ –m-TiO ₂ /FA _{0.8} MA _{0.16} Cs _{0.04} Pb(I _{0.84} Br _{0.16}) ₃ –RGO/spiro-OMeTAD/Au	T: 85 °C, RH: 40%	2.5	20	24.00	1.15	76.00	19.34	46	
Glass/FTO/TiO ₂ –RGO/TiO ₂ /MAPbI ₃ –RGO/spiro-OMeTAD/Ag	Mild humid, dark	50	60	22.90	1.01	72.00	16.50	107	
Glass/FTO/c-TiO ₂ /m-TiO ₂ /Al ₂ O ₃ /MAPbI _x Cl _{3-x} –Ag–RGO/spiro-OMeTAD/Au	T: 25–30 °C, RH: 45–57%	330	0	22.80	n/d	n/d	n/d	103	
Glass/FTO/c-TiO ₂ –RGO/m-TiO ₂ –MAPbI ₃ –RGO/spiro-OMeTAD/Au	n/d	n/d	n/d	16.50	0.84	58.30	9.30	34	
Glass/FTO/c-TiO ₂ /m-TiO ₂ –MAPbI _{3-x} Cl _x –RGO/spiro-OMeTAD/Au	n/d	n/d	n/d	22.30	0.93	74.00	15.30	209	
Glass/FTO/c-TiO ₂ /m-TiO ₂ /MAPbI _{3-x} Cl _x –N–RGO/spiro-OMeTAD/Au	n/d	n/d	n/d	21.80	1.15	74.00	18.73	138	
Glass/FTO/c-TiO ₂ /(FAPbI ₃) _{0.85} (MAPbBr ₃) _{0.15} –c-TiO ₂ –Li–RGO/spiro-OMeTAD/Au	n/d	n/d	n/d	21.98	1.11	80.00	19.54	124	
As HTL and electron blocking layer in n-i-p configuration									
Glass/FTO/TiO ₂ /MAPbI ₃ /GO/Au	Dark, no encapsulation	30	50	8.00	0.80	51.25	3.28	210	
Glass/FTO/c-TiO ₂ /m-TiO ₂ –graphene/MAPbI ₃ /GO/spiro-OMeTAD/Au	1 sun	0.67	67	22.48	1.08	75.12	18.19	211	
Glass/FTO/c-TiO ₂ /m-TiO ₂ –MAPbI _{3-x} Cl _x /GO/Cu–Bu–phthalocyanine/Au	n/d	n/d	n/d	20.90	1.04	66.00	14.40	212	
Glass/FTO/c-TiO ₂ /m-TiO ₂ /CsPbBr ₃ /polyvinyl acetate/GO/carbon	T: 25 °C, RH: 80%, no encapsulation	29	3	7.41	1.55	82.80	9.53	13	
Glass/FTO/c-TiO ₂ /m-TiO ₂ /MAPbI _{3-x} Cl _x /amine–GO/P3HT/Au	n/d	n/d	n/d	24.43	0.93	58.00	13.25	144	
Glass/FTO/TiO ₂ /MAPbI ₃ /RGO/spiro-OMeTAD/Au	Air	20	15	16.73	0.91	61.00	10.60	154	
Glass/FTO/Sr–TiO ₂ /Al ₂ O ₃ –graphene/NiO/MAPbI _{3-x} Cl _x /NiO–RGO/spiro-OMeTAD/Au	T: 25 °C	310	3	25.90	1.05	76.40	20.80	111	
In composites with HTL in n-i-p configuration									
Glass/FTO/SnO ₂ /FAMAI _{3-x} Br _x /spiro-OMeTAD–RGO/Au	Ambient	21	25	23.05	1.10	71.00	18.13	213	
Glass/FTO/c-TiO ₂ /m-TiO ₂ /MAPbI ₃ /spiro-OMeTAD–poly(methyl)methacrylate–RGO/Au	T: 35 °C, RH: 40%	42	7	22.60	1.01	68.00	15.70	214	
Glass/FTO/C ₆₀ /MAPbI ₃ /spiro-OMeTAD/Li–TFSI/P3HT–4-(hexyloxy)phenyl–RGO	n/d	n/d	n/d	20.00	0.87	55.00	10.00	215	
Lamination deposition of graphene derivative-based materials									
Glass/GO/MoO ₃ –PEDOT:PSS/MAPbI ₃ /C ₆₀ –BCP/LiF–Al	n/d	n/d	n/d	21.90	1.03	72.00	17.10	119	



Table 1 (Contd.)

Device structure	Conditions ^a	Stability		PCE decline (%)	J_{sc} (mA cm ⁻²)	V_{oc} (V)	FF (%)	Champion PCE (%)	Ref.
		Time (d)	n/d						
Glass/ITO/PEDOT:PSS–silver trifluoromethanesulfonate–GO/MAPbI _{3-x} Cl _x /PCBM/Au	n/d	n/d	n/d	19.18	0.88	70.51	11.90	116	
Glass/FTO/TiO ₂ /MAPbI ₃ /B-RGO/FTO	Ambient, dry box, room light, 10 RH: 60	n/d	16.74	0.88	60.00	8.96	159		
Electrospray deposition of GO layer									
Glass/ITO/TiO ₂ /MAPbI _{3-x} Cl _x /4-fluorophenyl-hydrazine hydrochloride–GO/spiro-OMeTAD/Au	RH: >50%, N ₂ -filled glove box, 1 sun	5	70	21.50	1.11	78.60	18.80	152 and 216	
Spray deposition of RGO layer									
Glass/FTO/c-TiO ₂ /nano-crystalline-TiO ₂ –MAPbI ₃ /RGO/Au	n/d	n/d	n/d	11.50	0.95	60.54	6.62	217	
Radio-frequency magnetron sputtering GO deposition									
Glass/ITO/GO/MAPbI ₃ /Ag	n/d	n/d	n/d	7.80	0.92	24.43	1.80	128	
Unspecified deposition methods for graphene-based materials									
Glass/FTO/GO/MAPbI ₃ /PCBM/ZnO/Al	Ambient, T: <30 °C	20	20	18.06	1.10	77.70	15.20	218	
Glass/bis(trifluoromethanesulfonyl-amide–GO/PEDOT:PSS:FAPbI _{3-x} Br _x /PCBM	T: 60 °C, RH: 30%, 1 sun	42	5	22.70	1.07	77.70	18.90	219	
Glass/ITO/RGO/MAPbI ₃ /PCBM/BCP/Ag	Ambient	6	38	15.40	0.98	71.60	10.80	220	
Glass/FTO/Zn-RGO/MAPbI ₃ /spiro-OMeTAD/Au	n/d	30	10	21.70	1.03	68.00	15.20	178	
Glass/FTO/spiro-bifluorene/MAPbI ₃ –GO/PCBM/bathocuproine/Au	Ambient	30	4	18.80	1.07	71.00	14.28	29	
Glass/FTO/TiO ₂ /MAPbI ₃ bilayer/GO	n/d	n/d	n/d	16.70	0.94	73.00	11.50	171	

^a n/d – no details, RH – relative humidity, AM – air mass.

substitute material to Pb-based PSCs.⁵⁶ Possible substitutes being considered include elements in the same periodic group, Sn and Ge; however, their toxicity, stability and band gap still need tailoring to achieve comparable absorption and PCEs.^{15,57} There is also the possibility of solvent-free approaches⁵⁸ although control over uniformity and morphology can prove difficult.

Another key aspect hindering commercialisation is the stability of the perovskite layer with, to date, the best-reported device lifetimes of approximately one year.^{14,28,59,60} The reviewed research has shown interesting stability profiles of PSCs in terms of shelf-life, continuous illumination, short circuit (increased accumulation of charge), bending cycles and defined relative humidity (RH) (Table 1). In-depth degradation studies have shown that ageing effects on PSCs are not homogeneous.⁶¹ Additionally, other comprehensive studies have monitored the degradation profiles of PSCs under operating conditions. They infer that stability is centred on phase separation (in hybrid halides), interface state(s), photosensitivity, and thermal, oxygen and moisture effects that cause perovskite degradation.^{9,61–66} Moisture (RH of >55%) catalyses perovskite degradation to a polar state *via* H⁺ extraction from the perovskite and facilitates the dissolution of water-soluble components as MAI, causing a colour change.^{15,67–69} Oxygen

detrimental effects occur when the molecules diffuse into the perovskite, and become trapped in the halide vacancies where interaction with an excited state forms reactive superoxide species.^{15,62,64,69} Ultraviolet and visible irradiation can also cause device degradation through accelerated ion migration.^{28,57,64,70–72} The blue to the ultraviolet region of the solar spectrum also triggers the dissolution of the perovskite organic component.⁷³ Perovskite degradation to PbI₂ can be thermally induced altering the physicochemical properties, such as a reduced optical absorption due to an enlarged band gap.^{15,63} Perovskite phase separations (cause photo-inactive states and current blocking) and phase transformations, such as FA_{0.9}Cs_{0.1}PbI₃ to FA_{>0.9}Cs_{<0.1}PbI₃, deteriorate PCE.^{63,64} Additionally, studies have shown that I₂ vapour does not only cause localised degradation but also induces decomposition of neighbouring perovskite regions during operational conditions.^{74,75} Interestingly, under illumination, I₂ is more easily released from PbI₂ than from perovskite,⁷⁴ meaning that once some perovskite decomposes to PbI₂ more problems are initiated.

In research laboratories, device fabrication is carried out in a glove box to ensure a controlled atmosphere, but this may not be easily achieved in large-scale industrial production. In addition, real-world PSCs will be applied in RH environments that exceed the laboratory average of 30–50%.⁷⁶ With the current



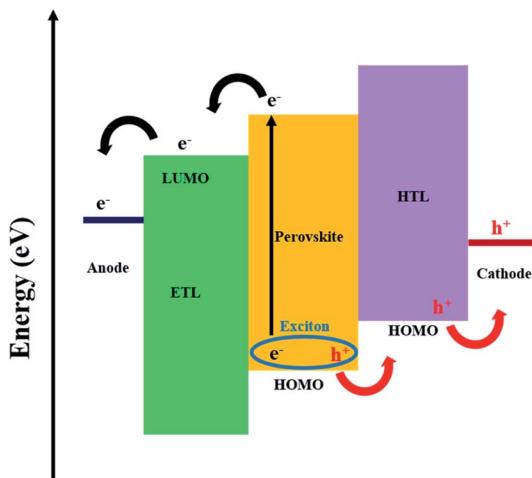


Fig. 1 Schematic energy level diagram of a conventional PSC.

state of the art, real-world devices will need to be fabricated in such a way as to limit humidity and oxygen exposure. Thus, more work is needed to develop new charge selective carriers, and engineer new perovskite materials, interfacial modifications, and novel configurations to achieve long-term stability.¹⁵

2.1 PSC configurations and components

The basic architectures of PSCs are either mesoporous (Fig. 2a and b) or planar (Fig. 2c and d). Mesoporous devices have a mesoporous (often TiO_2) layer on top of the compact ETL, whereas planar devices just have a compact ETL layer.⁷⁷⁻⁷⁹ Perovskites are suitable for planar structural solar cells due to

their ability to transport holes and electrons simultaneously, and have long exciton diffusion lengths (100–1000 nm) and favourable excited-state lifetimes (100 ns).^{56,80} PSC devices can be manufactured in both n-i-p (transparent conducting electrode (TCE)/ETL/perovskite/HTL/cathode (Fig. 2a and c))^{24,36,41} and p-i-n (TCE/HTL/perovskite/ETL/anode (Fig. 2b and d)) configurations.^{25,81,82} The difference between n-i-p and p-i-n is simply the position of the ETL and HTL relative to the transparent substrate and counter electrode (Fig. 2).²⁸

Studies on planar PSCs have highlighted the need for critical crystallisation and growth control to minimise morphological deviations (*i.e.*, defects; imperfect crystals and undesirable interfacial states) and recombination.^{83,84} The carrier lifetime, charge extraction and charge transfer issues that cause charge recombination and thus lower photocurrent in PSCs, particularly for planar architecture, have been linked to the ETL used.⁴⁴ Other common issues related to the introduction of charge transport layers in PSCs include the cost of materials, deterioration of electronic characteristics due to poor alignment of energy levels and charge extraction (slightly higher conduction band of ETL and lower HOMO of HTL than the valency band of perovskite, respectively, is needed to allow charge transport), slow electron mobility, high hydrophilicity of components causing poor chemical stability, and short-term stability and poor perovskite quality (creation of void infested morphology, poor crystallisation and crystal boundary state).^{21,85,86}

Generally, defects commonly found in perovskites can be categorised as intrinsic 0D (vacancy, interstitial and anti-site substitutions), 0D from impurities and 2D (surface defects, pinholes and grain boundaries).^{59,87} The perovskite grain boundaries are weaker than the interior for several reasons, such as the possible voids generated during precursor solidification, localised large volume and high stress intensity existence, and occurrence of amorphous boundary structures from Pb-X lattice distortions.⁸⁸ The interstitial and substitutional defects are deep level imperfections that lower perovskite stability and create charge traps.⁸⁹ Since perovskite film morphology affects stability and rate of degradation,⁹⁰ the quality of perovskites has been improved by optimisation of preparation conditions and compositional engineering, such as the use of graphene additives and selecting suitable starting materials.^{46,89,91,92} Additionally, defect passivation can reduce non-radiative recombination and energy losses, and readers are referred to detailed reviews by Gao *et al.*⁵⁹ and Zhang *et al.*¹⁴ on defect passivation. In brief, passivation is when a material's reactive/unstable component is made to become less affected, if not at all, by an external environment.^{59,68,90}

Graphene derivatives have shown potential in improving substrate coverage, defect passivation, and charge extraction/transport (as highlighted in relevant sections and Table 1). For instance, the inclusion of graphene oxide (GO) in the perovskite layer of a PSC lowered series resistance (R_s) from 27.8 to 15.2Ω .⁹³ This was ascribed to enhanced crystallinity of the perovskite in the composite. Additionally, surface potentials of a PSC were lowered with an increase of both J_{sc} and V_{oc} by lowering the HOMO level through the addition of polyaniline (PANI) to a GO/poly(3,4-ethylenedioxythiophene):polystyrene

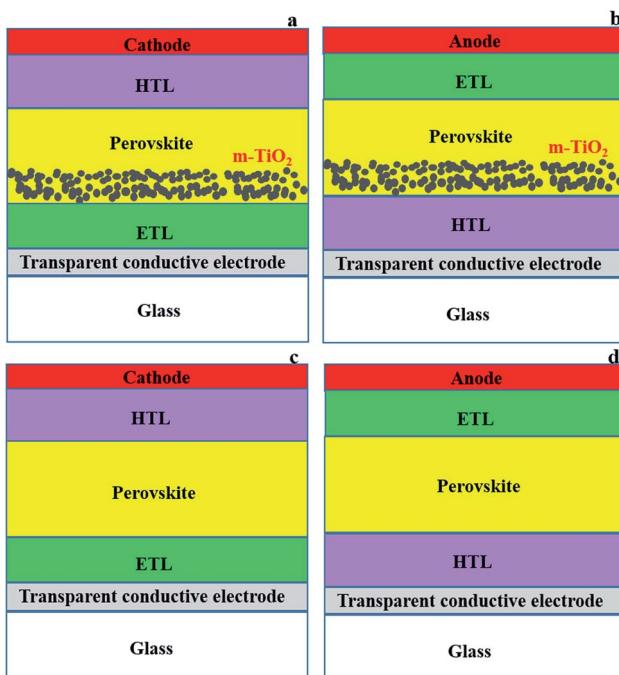


Fig. 2 Mesoporous (a) n-i-p and (b) p-i-n, and planar (c) n-i-p and (d) p-i-n PSC configurations.



sulfonate (PEDOT:PSS) composite.⁹⁴ Graphene derivatives have great potential in improving some of the most significant drawbacks of optimising PSCs. Despite this great potential, the use of graphene-based materials in PSCs is still at primitive stages;⁴² and, hence, needs constant review to reveal their potential holistically.

3. Application of graphene derivatives in perovskite solar cells

GO can be defined in simple terms as an oxide form of graphene that is chemically synthesised from graphite (Fig. 3a and b).^{95,96} Technically, the material obtained after partial removal of oxygen functionalities is termed reduced graphene oxide (RGO).⁹⁶ Furthermore, the development of several graphene derivatives has generated numerous and sometimes confusing terms, such as transferred, suspended, epitaxial grown, chemical vapour deposition (CVD)-grown, isolated-freestanding, and nanoplatelets, among others,^{97,98} which represent GO-based materials with different physicochemical properties. The major graphene derivative examples applicable in PSCs include GO (Fig. 3b), RGO (Fig. 3c), nitrogen-doped GO/RGO (N-GO/-

RGO), boron-doped GO/RGO (B-GO/-RGO) and metal oxide functionalised-GO/-RGO (Table 1). A high concentration of oxygen functionalities, particularly in GO, is usually associated with insulating characteristics, while a low oxygen concentration leads to high conductivity due to the restoration of the pi-system towards pure graphene. Hetero-atom doped graphene derivatives, such as those that are N-doped, also have enhanced conducting behaviours.

A comprehensive summary of the use of graphene derivatives in PSCs is presented in Table 1. It shows that there were few reports on graphene-based PSCs in the last decade, probably due to the drawbacks that generally evolve around high-temperature synthesis requirements.^{21,47} Graphene derivatives are mostly applied as components of the HTL, ETL, blocking layer, and conducting electrode and, rarely, in the active layer of PSCs. The PCE values of graphene-based PSCs have grown from about 6.6% to the current value of approximately 17%.⁹⁹ Although the champion PCE of graphene-based PSCs is currently lower than the certified champion PCE value for PSCs (of 25.5%),³³ the relative increase in performance of graphene derivative-based devices from their first use, has been attributed to several factors that are inclusive of chemical inertness,

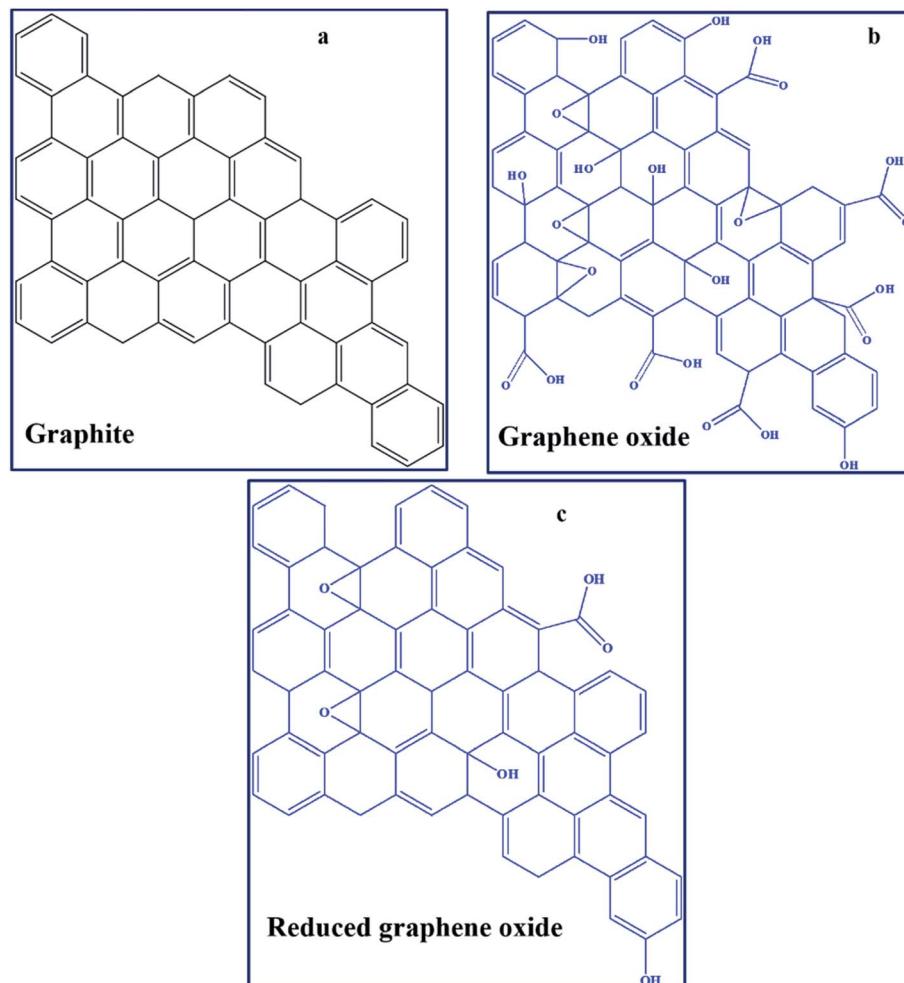


Fig. 3 Basic chemical structures of (a) graphite, (b) GO and (c) RGO.



increased charge transfer pathways, and surface area. For instance, a relatively large surface area of $202\text{ m}^2\text{ g}^{-1}$ in a graphene derivative-based PSC (graphene derivative applied on top of mesoporous ZrO_2 and TiO_2) created a high number of interconnected channels that allowed perovskite precursor penetration and decreased the photoluminescence (PL) lifetime (from 1.03 to 0.698 ns).¹⁰⁰

The use of graphene grown by CVD in PSCs potentially introduces high costs for large-scale manufacturing because of complicated transfer steps of graphene onto the target substrate. In addition, it is associated with both poor contact and minimum film thickness control.²¹ Hence, GO is more appropriate for large-scale PSC development than graphene because solution processable GO synthesised *via* several chemical exfoliation methods⁹⁶ can disperse well in several solvents compatible with PSC fabrication, and allows for potential further functionalisation.

Since the mechanism of PSCs is inclined to the charge transport layer used for quenching charge carriers,²⁵ fast charge decay is affected by fast transport of free carriers from the perovskites to respective contacts.¹⁰¹ The electron and hole mobility need to be balanced in PSCs, and this can be attained through the synthesis of composites, such as polymer-perovskite,^{102,103} metal oxide-perovskite^{104,105} and graphene derivative-perovskite composites.^{46,106} For instance, GO-perovskite composites were reported to increase charge separation (PL coefficient due to non-radiative surface-state relaxation of perovskite was one-tenth of that of the composites), charge mobility (from 29 to $35\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$) and recombination resistance (from 8865 to $99\text{ 978 }\Omega\text{ cm}^2$).⁸⁰ Also, the addition of RGO to mesoporous TiO_2 [(m- TiO_2)-RGO] was reported to double the electron diffusion coefficient from that of a m- TiO_2 ETL.²⁶ In another study, the electron mobility of m- TiO_2 was raised from 2.5×10^{-7} to $4.2 \times 10^{-7}\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ in a RGO-m- TiO_2 ETL.¹⁰⁷ Doping of PEDOT:PSS with GO was reported to improve the hole mobility from 5.55×10^{-5} to $1.57 \times 10^{-4}\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$.¹⁰⁸ These studies demonstrate the potential of graphene derivatives in promoting charge mobility *via* the provision of additional pathways for effective excited-state charge carriers and charge transport away from the PSC active layer (Table 1). This is partly due to the ability of graphene derivatives to passivate surface/interfacial defects between perovskites and charge transport materials,¹⁰⁹ and lower defect concentrations by promoting the uniform growth of larger crystals (from 1024 nm up to 1250 nm).^{36,107} Graphene derivatives can participate in defect passivation at both exterior and interior film surfaces through bonding of their functionalities ($-\text{NH}_2$, $-\text{OH}$, $-\text{C}=\text{O}$) to uncoordinated ions on the perovskite surface.^{109,110} Hence, graphene derivatives, such as GO, have lucrative research prospects for defect passivation in PSCs.

The conventional and inverted planar architectures of graphene derivative-based PSCs have achieved PCE values of 18.2 and 13.8%, respectively.⁴⁴ In recent times, other positive prospects have shown the inclusion of NiO-RGO in a glass/FTO/Sr-TiO₂/Al₂O₃-graphene/NiO/perovskite/NiO-RGO/spiro-OMeTAD/Au configuration, achieving 95% of the theoretical photocurrent density of 27.2 mA cm^{-2} (Table 1).¹¹¹ This was

driven by the ability of NiO-RGO to lower the trap state concentration of electrons and holes (from 7.09×10^{15} to $3.59 \times 10^{15}\text{ cm}^{-3}$ and 9.57×10^{16} to $2.71 \times 10^{16}\text{ cm}^{-3}$, respectively), and facilitate fast charge mobilities (from 4.12×10^{-2} and $1.12 \times 10^{-1}\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ to 5.05×10^{-1} and $1.33 \times 10^{-1}\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$, for holes and electrons, respectively). This highlights the suitability and potential benefits of using RGO composites in PSC technology.

Other important aspects of a PSC device are linked to conductivity and band alignment of constituent materials. It is difficult to balance the band alignment effects of GO with conductivity properties towards effective charge separation; hence, more research is needed. While the high work function (WF) effects of graphene derivatives promote better band alignment, ohmic contact and building of a potential difference towards higher V_{oc} , an increased conductivity leads to a high charge collection efficiency and J_{sc} .^{21,112} On the one hand, the valence band position of the HTL should align with the HOMO of the perovskite layer (Fig. 1).^{56,99} For example, the synergy between GO and polyvinyl acetate was reported to lower the energy level difference between the valence band of the perovskite and the WF of a carbon electrode (from 0.56 to 0.28 eV), and increase hole extraction (shortened PL lifetime from 0.36 to 0.29 ns) between polyvinyl acetate and carbon in a FTO/compact TiO_2 (c- TiO_2)/m- TiO_2 /perovskite/polyvinyl acetate/GO/carbon cell.¹³ On the other hand, the LUMO of the perovskite should align with the LUMO of the ETL (Fig. 1). Theoretically, graphene has a WF of 4.5 eV,¹¹² while derivatisation to GO and RGO was associated with WF modulation into the range of 4.35–5.28 eV⁴⁶ and 4.9–5.0 eV,^{112–114} respectively. The WF of GO can also be tailored by heteroatom functionalisation (by replacing oxygen moieties), by reacting GO with molecules such as NH_3 ⁸⁵ and doping^{21,115} with elements such as B⁹⁹ and F.¹¹⁴ For example, silver trifluoromethane sulfonate doped GO achieved a lowered R_s (from 16.44 to $12.11\text{ }\Omega\text{ cm}^2$) and electron/hole recombination due to WF modulation.¹¹⁶ Mann *et al.*¹¹⁷ also reported RGO WF modulation to 5.34 eV *via* sulfur doping that improved functionality of PEDOT:PSS to achieve a 19% increase in the PCE and improved stability (stable PCE after 10 days for sulfur-RGO/PEDOT:PSS *versus* continuous decline for 30 days in control PEDOT:PSS devices). The graphene derivative dopant must increase performance without compromising other parameters, such as transmittance and charge extraction. For illustration, Kim *et al.*¹¹⁸ altered sheet sizes of GO by sulfur-doping, and this influenced the extraction of charge carriers in a PSC (PCE increased with decrease in size of GO with a corresponding increase in PL lifetime: 6.55, 8.87 and 11.3 ns for <1, 1–16 and >25 μm^2 sizes, respectively), highlighting the importance of optimising sheet sizes. Composite synthesis was also reported to modulate the WF; for example, MoO_3 and Ag raised the WF of GO from 4.23 to 4.71 eV¹¹⁹ and 4.95 eV,¹⁰³ respectively. Another reported functional composite example is GO-Li, with a lowered GO WF of 4.3 eV (from 4.9 eV).¹²⁰ This was rationalised by the ability of Li to donate a valence electron to GO and the associated induction of dipoles by Li^+ ; therefore, the Fermi level shifts towards vacuum and the WF is minimised.



GO is transformed to a semiconducting character by tuning the associated WF through reduction.^{95,121} Whilst it is a good fabrication strategy to utilise the high dispersibility of GO in promoting substrate coverage with perovskite, the subsequent reduction process may introduce complications in PSCs. On the other hand, despite aggregation effects and poor dispersionability from prior GO reduction, fabrication may require the use of both surfactants (such as sodium dodecylbenzene sulfonate)²² and chemical reductants (such as hydrazine hydrate),^{22,121} which may be difficult to handle at large-scale due to their toxic and explosive properties. A moderate annealing temperature in air (usually 200 °C), UV radiation and chemical treatment are possible routes to reduce GO towards improved device conductivity.^{24,56,122,123} Annealing graphene derivatives at high temperatures (above 500 °C) for PSCs is not suitable for most flexible substrates and perovskites (stability and contact issues). Typical contact problems can be improved by using volatile pore-forming agents that evaporate upon thermal treatment, leaving voids occupied by the perovskite.²¹

The effective role of graphene derivatives in PSCs still needs further in-depth understanding. For example, Kim *et al.*⁴⁶ studied the effect of the oxygen content of GO (7, 10 and 18%) used as an additive to the HTL (spiro-OMeTAD) of PSCs and comprehended that 10% oxygen content was the optimum with a negligible drop in PCE after 18 days of storage under a RH of 20%. Cho *et al.*¹²⁴ investigated the most effective location of RGO in a PSC and obtained the best results when it was used as an ETL component (3% better than control, while PCEs of the rest were below that of the control). A similar study reported a 20% PCE improvement through a simultaneous inclusion of RGO in the ETL and active layers of a PSC.¹⁰⁷ From the WF discussion, it is possible that the oxygen concentration influences RGO performance in PSCs. This view is supported by density functional theory (DFT) calculations that showed that the epoxy oxygen concentration tunes the charge extraction ability as follows: if <33% then both holes and electrons can be extracted, for the 30–66% range electron transport is permissible, but if >60% no extraction is conceivable.¹²⁵ However, since it is rare to find only one particular type of oxygen species in a graphene derivative after synthesis, it is not easy to attain such tailoring due to the presence of other moieties.

Here we detail the suitability of graphene derivatives in PSCs as ETL,^{115,126,127} HTL,^{56,81,82,115,126,128} buffer and active layer additives,^{56,81} and transparent conducting electrodes (TCE) (Table 1).^{22,81,122} The widespread roles of graphene-based derivatives in PSCs affords advantages that most likely arise from favourable interactions with other components, configurations, experimental conditions, unpredictable perovskite nature, and varying the oxygen species and their concentration. However, due to the large volume requirement, the most likely drawbacks are linked to solvents and large-scale GO-based PSC production.²¹

3.1 Transparent conducting electrode

The role of a TCE, commonly designed to be the layer facing the light source, is to collect electrons from a semiconductor and

transfer them to the external circuit. The most common TCEs are fabricated from indium- and fluorine-doped tin oxide (ITO and FTO). The problems associated with these commonly used TCEs include high fabrication costs (contributes 58–73.9% of the material costs)^{7,129} and energy consumption, natural brittleness, and the scarcity of indium (in the case of ITO glass), which is estimated to be 0.05 ppm and 0.072 ppm in the continental and oceanic crust, respectively.¹³⁰

Theoretical simulations conducted with COMSOL and SCAPS-1D software have predicted graphene-based materials to be effective as either top or bottom TCEs in PSCs as they offer better heat dissipation and stability than their metallic counterparts.^{131,132} The thermal stability of RGO promotes heat dissipation within the PSC,⁹⁵ hence, reducing the risk of thermally-induced device degradation. While the use of pristine graphene as both front and back electrodes can work,^{78,133,134} its intrinsic low wettability, due to its hydrophobic nature, causes the formation of poor perovskite films, limiting its usage. Other graphene derivatives, such as GO and RGO, are suitable PSC TCE alternatives due to their good mechanical strength (Young modulus of 1 TPa), ability to facilitate heat dissipation, and lower perovskite degradation and ion migration rates at elevated temperatures. P-Type graphene derivative-based TCEs, such as GO, are relatively stable in air and therefore easily produced.¹³⁵ GO reduced by chemical or laser means is suitable for TCEs fabricated by spin-coating and result in PSCs with low charge impedances.

Despite the usefulness of graphene-based derivatives as TCEs, their practical application has been greater in other PSC layers, as reviewed in the subsequent sections. This is probably due to limitations in light transmittance to the perovskite layer, because graphene derivatives absorb visible light, and difficulties in deposition onto substrates. The drawbacks of replacing common TCEs in PSCs with solution-processed GO include lower conductivities and PCEs.⁹⁵

Graphene derivatives are appropriate for PSCs that use transparent and flexible substrates such as cellulose and plastics.^{21,81} Examples of graphene derivative-based TCEs in a flexible PSC are single-walled carbon nanotubes (SWCNTs)-RGO,⁸¹ SnO₂-GO,¹³⁶ Au-Cl₃-doped graphene¹³⁷ (with the use of 3-aminopropyl-triethoxy silane as adhesion promoter), and Ag nanowires-GO (Table 1).¹²² Despite the high prospects of Ag nanowires in replacing ITO and FTO due to high transmittance and low sheet resistance, they suffer from high costs and poor chemical stability when exposed to halides. Ag nanowire-GO composites are able to improve the chemical stability in flexible PSCs.¹²² However, the hydrophobic nature of the Ag nanowire-GO composite can also introduce problems in the HTL and perovskite film during fabrication.¹²²

Although graphene derivative-based TCEs are more appropriate for inverted PSC types, applications in conventional devices have also been reported. As an illustration, Kim *et al.*¹³⁵ reported a Ag nanowire-doped graphene-based TCE for n-i-p applications that exhibited a PCE of 15.8 and 13.5% for rigid and flexible PSCs, respectively. Hence, this could possibly be extended to GO and RGO in which they can be additional



components or substitutes for traditional materials either as transparent front electrodes or electron acceptors.

3.2 Active layer

The role of an active layer in a solar cell is to harness sunlight and, in turn, create charge carriers. Several graphene derivative-perovskite composites for use in the PSC active layer have been reported for a variety of reasons (Table 1). For instance, graphene derivatives such as N-RGO-perovskite composites have two possible effects on PSCs; firstly, by increasing the grain size through a decelerated crystallisation step, both the J_{sc} and FF are increased,¹³⁸ and, secondly, the composites retard the rate of recombination, thereby increasing the V_{oc} . Other typical examples are the GO-modified perovskite materials reported as sensitizers that improved hole mobility due to the enhanced perovskite quality arising from a smaller number of grain boundaries (grain size increased from 150 to 200 nm) and improved uniformity.²⁹ The optimum performance of the composites was subject to the wt% of GO (optimum = 0.5 wt%). This means that even if the inclusion of graphene derivatives has benefits, maximum gains to overall PSC performance require optimisation of the amounts of components to avoid the formation of inhomogeneous films that culminate in a disrupted crystal quality. In summary, solution-processed graphene derivatives have the potential to improve crystallisation, morphology and charge transport in PSCs.

3.3 Blocking layer

Buffering in PSCs is important in several ways; for example, in minimisation of the reaction between extracted charges and X^- .¹¹⁰ Buffering layers are utilised in blocking holes and electrons from collecting at the anode and cathode, respectively. This ensures a balanced charge extraction towards the appropriate electrodes.⁸¹ Common buffering layers in PSCs are TiO_x and PEDOT:PSS. Although the PSS component of PEDOT:PSS lowers the overall conductivity,¹³⁹ its hygroscopic nature is manipulated for dispersion in water as a counter ion.¹⁰¹ The acidic and hygroscopic nature of PEDOT:PSS,^{101,140} as well as moisture sensitivity of TiO_x , are some of the common setbacks of current blocking materials. Moisture in PSCs will not only deteriorate the perovskite at atmospheric conditions but also reduce electrode conductivity.¹⁴¹ Additionally, PEDOT:PSS has an intrinsic inefficient electron blocking ability.³⁶

GO generates structural inhomogeneities through its high oxygen functional group concentrations that can aid retardation of electron/hole recombination by effecting low electron mobility rates.^{140,142} For example, the wider band gap of GO induces an effective blocking layer in PEDOT:PSS-GO composites (Table 1).⁵⁶ The PEDOT:PSS-GO composite is currently among the most effective materials that can act as an electron blocking layer and HTL between the active layer and electrodes in order to minimise current leakage and electron recombination. Although improving the PCE with composites of graphene derivatives for use as HTLs and ETLs is gaining momentum, their use as a blocking layer is still less common than their TCE applications.¹⁴³ Attributes of graphene derivatives as buffer materials include solution

processability, excellent stability, low fabrication costs and their diverse WF tunability. The WF modulation of GO through reduction is a feasible way to engineer energy barriers between the HTL and active layer.¹⁴³ Heat and moisture barriers between the perovskite and the electrodes in a PSC can be created by inserting graphene derivatives.¹⁴

Other graphene derivatisation examples for blocking purposes include SWCNTs- and amine-modified GO (Table 1).¹⁴⁴ A fluorinated ($-CF_3$)-GO was reported to promote both oxidation resistance and moisture adsorption, thereby inducing better device stability (Table 1).⁵¹ Another example is a poly(methyl methacrylate) (PMMA)-GO composite buffer that was reported to block electrons and increase carrier transport selectivity through shunt resistance (R_{sh}) enhancement and a decrease in R_s , respectively.¹⁴⁵ Also, the strong chemical interactions between NiO, RGO and the perovskite in a NiO-RGO blocking layer were envisaged to boost stability against reactions with air and water (Table 1).¹¹¹

3.4 Hole transport layer

The role of HTLs is to extract holes generated in the perovskite layer.¹⁴⁶ In some instances, PSCs have been reported to function without a HTL but with just an insulator that separates the two electrodes.^{21,147-149} In such cases, the perovskites will act as a sensitizer and an ambipolar charge transporter.^{138,150,151} Exclusion of a HTL is beneficial as it can reduce material use and fabrication costs and aids in resolving stability issues such as those from increased moisture effects that may be introduced by hygroscopic HTLs such as PSS (a component of PEDOT:PSS). However, the HTL can improve the PCE by reducing recombination rates and enhancing V_{oc} . The V_{oc} in PSCs is mostly influenced by recombination at interfaces. This effect is large in HTL-free devices due to the longer-lived duration of charges at interfaces. Another driver for HTL use is that the surface morphology of the HTL positively influences electrical properties such as R_s and R_{sh} . This is important because a large R_{sh} infers small charge carrier recombination in the active layer and high selectivity of charge collection.^{101,152} An efficient hole selective collection will occur when current leakage in the reverse bias is decreased.¹⁵²

To be a suitable HTL in PSCs, the material requires an appropriate energy level, low deep-trap state density, sufficient charge extraction rate and transfer capabilities.⁸³ A commonly used HTL for the p-i-n configuration is PEDOT:PSS due to its tuneable conductivity and high optical transmittance within the visible region.^{82,115} Also, since the WF of GO (4.9 eV) and that of PEDOT:PSS (5.1 eV) align well, GO is able to decrease the associated R_s .^{56,140} Copper phthalocyanine²⁷ and spiro-OMeTAD^{142,145,148} are also well reported functional HTLs. Owing to a compatible energy level with perovskites, spiro-OMeTAD is common in Au-based PSCs though it requires a complicated multi-step synthesis, is expensive and associated with low intrinsic charge carrier mobility, and thus requires dopants to perform well.^{12,15,21,54}

GO functions as a hole acceptor and transporter,^{80,153} but has a lower conductivity than RGO, rendering it an inferior



transporter.¹⁵⁴ The GO HTL can tailor optical properties,¹⁵⁵ crystallisation and morphology of perovskites.¹⁴⁰ A uniform, compact and void-free GO layer is important for device performance, but some studies have encountered difficulties in achieving this with GO.^{56,115} Unlike N-RGO, RGO has been reported in some instances to cause deterioration of perovskite morphology.⁹⁵ Cracks and voids in the perovskite layer are recombination centres with poor uniformity leading to increased shunt paths, current leakages, and low voltage.^{21,115,156} This phenomenon is subject to the deposition method and is a signal of the importance of uniform layers throughout a PSC. Furthermore, inhomogeneity of GO layers on TCEs produces non-uniform electrical characteristics due to either direct contact between the perovskite and TCE or insulating behaviour that results in poor hole transport.

Lee *et al.*⁵⁶ overcame this shortfall by a repeated spin-coating of GO and PEDOT:PSS to reduce surface roughness. Sequential solution deposition has achieved decent, repeatable, and easily controlled deposition, proving better than both vacuum deposition and vapour-assisted solution-processing.^{142,157} Nouri *et al.*¹¹⁵ achieved a uniform thin film by first cleaning FTO with ethanol to increase its hydrophilicity, and then spin-coating twice with GO dispersed in isopropanol at 200 rpm (PCE increased from 4.4 to 10.2%). Feng *et al.*⁸⁵ reported PEDOT:PSS/GO-NH₃ as a HTL with improved optical transparency, texture and complete substrate coverage morphology. NH₃ treatment of GO used as a HTL was also reported to increase the PCE of GO-based inverted PSCs, by 12%⁸⁵ and 18%,¹⁵⁸ and the PSC device was more stable than that from pristine GO due to advanced V_{oc} and FF, and preferred crystal orientations, respectively (Table 1). The enhancement was ascribed to a decrease in current leakage and increased V_{oc} from the compact perovskite morphology induced by NH₃-GO. Selvakumar *et al.*¹⁵⁹ reported an increase of both mobilities (from 0.595 to 4.850 cm² V⁻¹ s⁻¹) and R_{sh} (from 152 to 10 669 Ω cm⁻²) but a decrease in conductivity (from 4820 to 1521 S cm⁻¹) upon B-doping of GO. Habib *et al.*¹⁶⁰ reported an optimised ratio of PANI : GO of 1 : 0.5 with a sheet resistance of 0.067 Ω cm⁻² and PCE boost of 125% due to reduction of current leakage from smooth surfaces (root mean square of 12.65 nm) in the bulky materials.

The lifetime of a PSC is subject to pH, humidity level, permeability, density and hydrophobicity of the HTL.^{38,161} This means that PEDOT:PSS limits lifetime and cell performance due to its hydroscopic nature.¹⁰¹ Encapsulation improves device stability and can maintain the integrity of PSC layers over time.⁷⁴ The use of graphene derivatives as encapsulant materials has the potential to contribute significantly to enhancing perovskite device longevity. An illustration of this is the reported lowering of the hydroscopic character of PEDOT:PSS through the synthesis of an annealed composite with GO.¹⁶² However, depending on their chemical nature, graphene derivatives may pose drawbacks in their long-term utilisation as HTLs since they may introduce considerable moisture in the active layer. This is detrimental to PSC devices because moisture dissolves the methylamine and/or hydrates the perovskite and causes it to decompose to PbI₂ in a few hours or days when exposed to air and moisture.^{38,85,101} Possible ways to keep

moisture out of the perovskite layer and thereby minimise perovskite degradation include employing a hydrophobic HTL such as hydrophobic RGO. Another graphene derivative-based HTL with minimal hydrophilicity is thiolated GO as in the perthiolated trisulfur-annulated hexa-*peri*-hexabenzocoronene-GO composite form.⁸³ Additionally, a defect-free monolayer of some graphene derivatives is impermeable to all gases, preventing O₂ diffusion into the perovskite.¹⁶³ For instance, F-RGO HTL was able to minimise moisture and O₂ attack on the perovskite layer resulting in 72% PCE retention after 30 days (the control PEDOT:PSS-based device malfunctioned in 9 days).¹⁴⁶ Similar reports detail an isopropanol ultrasonically-treated graphene-based encapsulant enhancing the stability of a PSC by 90% within 60 h by reducing oxygen oxidation and moisture attack.¹⁶⁴ In addition, the high thermal conductivity of graphene helps with thermal stress on PSCs, leading to additional stability benefits.^{95,109} A graphene derivative–carbon paste has been reported as a suitable encapsulant of PSC modules of areas 0.09, 1 and 4 cm² that achieved champion PCEs of 15.81%, 14.06% and 13.86%, respectively.¹⁶⁵ Additionally, graphene-based encapsulants are compatible with roll-to-roll fabrication through a dry transfer method that has attained PCE retention of 82.4% after aging in ambient air for 3700 h.¹⁶³ This demonstrates that graphene derivatives, as encapsulants, are beneficial and conducive to the future scaling-up of PSCs. Other hydrophobic materials, such as a cross-linked compacted poly(methyl)methacrylate (PMMA) layer on top of the HTL,¹⁴⁵ have been used to modify the perovskite surface and seal the active layer from atmospheric effects and therefore increase stability. The compatibility of graphene derivatives with common PSC encapsulants, such as PMMA, has been illustrated. For example, a PSC encapsulated with a PMMA–RGO composite displayed excellent performance under operational conditions because the OH⁻ moieties on RGO interact with the C=O on PMMA.^{95,109} However, strict and effective encapsulation may introduce not only additional cost (60% more)^{38,166} but also hinder optical transparency.¹⁶⁷ Hence, a careful cost-effective sealing innovation needs to be sought for moisture insensitive designs involving graphene derivative HTLs. In a study to compare stability and PCE upon inclusion of GO in PSC devices, the glass/FTO/NiO/GO/perovskite/GO/TiO_x/Al cell structure was more stable but with a 25% lower PCE than the glass/FTO/PEDOT:PSS/MAPbI_{3-x}Cl_x/PCBM/Al configuration.¹⁶⁸ Hence, further understanding of the degradation mechanism through focused studies on individual PSC layers rather than the overall PSC device may provide solutions to both stability and PCE enhancement.¹⁶⁹

For HTL purposes, graphene can be chemically modified as an interlayer material to simplify the electrode solution coating process by improving wettability during the coating process. The enhanced wettability from derivatized graphene reduces the nucleation barrier to form a better quality perovskite layer without pinholes.¹⁰¹ The perovskite quality boost occurs *via* heterogeneous nucleation (nucleation at foreign nuclei or surface), and this process is influenced by the surface or interface contact energy (surface chemistry and morphology effects).¹⁷⁰ This infers that improved wettability (small contact



angle) is associated with a small contact energy barrier and, together with a uniform large surface area, promotes continuous crystal growth. For illustration, GO was modified with 4-fluoro phenyl-hydrazine hydrochloride (4FPH) to form a 4FPH-GO composite with increased passivation of trap states close to the perovskite surface (disorder parameter that represents broadness of trap state densities decreased from 151 to 81 meV) and improved charge diffusion length (from 400 to 700 nm) for effective hole collection, and hydrophilicity for easy coating (Table 1).¹⁵² Furthermore, the use of GO has been shown to decrease the contact angle between spiro-OMeTAD and the perovskite (from 21.6 to 2.5°¹⁰¹ and from 13.5 to 0°¹⁴²), leading to better spreading of the HTL on the perovskite.

HTL surface properties, structure and optical characteristics also influence film quality and suitability in effective light-harnessing. The optical properties of a HTL are important because it is the first component to significantly interact with light in the inverted configuration.⁸⁰ Hence, GO absorption of 2.3% of incident light in the 200–800 nm range may affect the overall PCE.²⁶ Hence, a high concentration of graphene derivative additives introduced into the HTL and poor dispersion can cause reduced film quality (light transmittance) and ultimately lower PCEs.^{21,80} This aspect is closely related to the film thickness effect as reported with functionalised GO and composites.^{36,135,152,153} Despite the possible increase of R_s due to the high oxygen concentration in GO,¹⁰¹ the quenching effect has also been reported to increase with composition.²⁵ For illustration, a 6.7% PCE was reported for a single GO layer (80% quenching effect improvement) in a PSC device, whilst a bilayer recorded 11.5% (91% quenching effect boost) due to the formation of an effective Schottky barrier, minimising recombination effects.¹⁷¹ In brief, there is a trade-off between optical transmittance and electrical resistance. On the one side, the electrical resistance is lowered by thick graphene derivative sheets, whilst on the other side, thin films have higher optical transmittance.^{22,120,140,143} Resistivity of graphene was shown to decrease from 1050 to 140 $\Omega \text{ sq}^{-1}$ when the thickness was increased from 1 to 2 layers.¹²⁷ SCAPS-1D simulations have also shown the effect of thickness on V_{oc} up to 1 V and an optimum thickness of 30 nm, and a decline in FF with thickness increase.¹³² Since several works have reported V_{oc} above 1 V, this effect is likely to be dependent on the material properties (Table 1).

To sum up, the review of graphene derivatives in the context of HTLs shows their potential in increasing hole transport and their electron blocking effect, thereby improving the electrical properties of the device (conductivity and R_{sh}), FF and V_{oc} .³⁶ Additionally, graphene derivatives have generally been able to improve the perovskite crystallisation process producing larger crystals, TCE surface coverage and in-plane orientation towards effective hole extraction.^{25,101}

3.5 Electron transport layer

The function of an ETL in a PSC is to extract and transport generated electrons away from the perovskite active layer. The three common classes of ETLs for PSCs are metal oxides, n-type conjugated polymers, and fullerenes.¹¹⁵ A typical example of

fullerenes for ETL is phenyl-C₆₁-butyric acid methyl ester (PCBM). PCBM facilitates a fast charge transfer from the perovskite (approx. 0.4 ns), and its hydrophobic nature enhances PSC stability.¹⁴⁰ Due to the low cost of graphene derivatives and their ability to introduce additional percolation channels for electron transport, solution-processed graphene derivatives have the potential to advantageously replace the popular PCBM.^{56,81} Hence, the basis of adding graphene derivatives to the active layer as ternary components of polymers and fullerenes is to ultimately increase carrier mobility.

Although a report indicates that PSCs can also function without an ETL (with PCE of 10%),¹⁷² ETLs are important in lowering interfacial energy barriers between electrodes and the perovskite active layer. Hence, electron/hole recombination rates are lowered by the ETL through electron extraction and transfer enhancement.¹⁷³ The tuning of the ETL surface energy is a feasible control protocol for wettability, recrystallization and nano-structural parameters of perovskites.¹⁷³ With reference to the PSC structure, the n-i-p type is largely influenced by the electrical, optical and structural features of an ETL.³⁵ Common ETLs for the n-i-p structure are oxides, such as TiO₂,³⁴ SnO₂ and ZnO.^{35,44,174}

The processing temperature of an ETL during PSC fabrication influences device performance. For instance, both c- and m-TiO₂ require a high sintering temperature (*ca.* 450 °C) for optimum crystallinity and charge mobility.^{21,37,175} A high sintering temperature hinders applicability in widespread areas, such as wearable and flexible electronics. ZnO is a good alternative because it requires a low sintering temperature and time; however, its non-optimised dimensions (diameter: ~80 nm and length: ~1 μm) negatively affect perovskite crystal growth in PSCs.⁷⁷ A m-TiO₂ layer may hinder electron transport due to roughened grain boundary scattering.^{26,176} Ternary n-type semiconducting metal oxides with similar characteristics as TiO₂, such as high Hall electron mobility and small effective mass, are also effective ETLs that have the potential to replace TiO₂ in PSCs.¹⁵³ For instance, the TiO₂ hysteresis shortfall has been resolved by the use of SnO₂ despite its low thermal stability, and non-optimal optical and electrical properties.^{35,84}

Interestingly, graphene derivatives are compatible with metal oxides that function in ETLs to form meso-superstructured PSCs with preferred properties (Table 1),²⁵ such as increased semiconducting properties,^{121,173} device stability and charge extraction.²⁷ For illustration, the inclusion of RGO within TiO₂ layers of PSCs produced a PCE boost of 21%.²⁷ A typical example is the overturning of the slow electron diffusion rate limitation in TiO₂, due to high resistance and low charge collection efficiency, by the use of GO as an interfacial layer between the TCE and TiO₂.^{26,34,37,124} Although all values were lower than the FTO control (J_{sc} of 17.49 mA cm⁻² and FF of 63%), altering the % of GO in c-TiO₂ has also been reported to proportionally enhance J_{sc} and FF due to increased electron transport through the creation of thermodynamically favourable energy paths and lowered recombination.²² Xie *et al.*¹⁷⁷ reported a V_{oc} boost of up to 1.12 V through RGO inclusion in a MoO₃ ETL (Table 1). Additionally, N-GO was used to reduce oxygen vacancies in SnO₂ ETLs as a control measure (5



volume% lowered O 1s peak in X-ray photoelectron spectrum and improved PCE by 8%, Table 1).³⁵ In another study, a RGO/Zn₂SnO₄ nanofiber scaffold was reported to effectively reduce recombination relative to pristine Zn₂SnO₄ (slow PL lifetime was reduced from ~17 to 4 ns, Table 1).¹⁵³ Also, ZnO-RGO quantum dots were reported as a PSC ETL, and the RGO passivated the ZnO effectively, improving charge carrier extraction and the stability of the perovskite due to a decline of the OH⁻ concentration on the surface (10% PCE drop *versus* 90% of control after one month).¹⁷⁸ In another example, Li neutralised GO was reported to increase electron extraction from the perovskite as an ETL.¹²⁰ In short, the electron conductivity of an ETL plays an important role towards effective charge extraction from active layers in PSCs, and compositional effects must be investigated in each case when graphene derivatives are involved.

3.6 Tandem-structured devices

Tandem solar cells (TSCs) are stacked narrow band gap and wide band gap subcells with high potential to suppress the Shockley–Queisser limitation of single solar cells and increase V_{oc} and J_{sc} (Fig. 4).^{165,179,180} This means only photons with energy $>$ band gap of a semiconductor are absorbed and not all absorbed photons can be converted to electricity due to thermalisation of charge carriers.¹⁸¹ Leading tandem structures involving perovskites are perovskite/perovskite (17% share), perovskite/CIGS (25% share) and perovskite/Si (58% share).¹⁷⁹ Perovskite/Si TSCs have reached a certified value of 29.5%.³³ Illustratively, graphene has been shown as a suitable material for TCEs (PCE of 8.3%),¹⁸² electrodes (PCE of 18.1%),¹⁸³ and as an additive to the HTL (PCE of 6.02%)¹⁸² of TSCs. Graphene derivatives can hypothetically slow down charge thermalisation processes.¹⁶⁵ However, there are currently scarce applications of graphene derivatives in TSCs. Theoretically, since a tandem structure will require an efficient PSC, the

demonstrated positive effects of graphene derivatives in various layers of PSCs (Table 1) infers that graphene derivatives are promising in TSCs. Hence, the progress of integrating graphene derivatives in TSCs is dependent on their status in PSCs. Since TSCs require highly conductive intermediate layers with low visible light absorption capabilities to separate subcells,^{127,182–184} the future application of graphene derivatives in TSCs must pay much attention to conductivity and transparency dynamics.

4. Deposition techniques for graphene derivatives and stability of PSCs

Most graphene derivatives utilised in PSC fabrication, as summarised in Table 1, are synthesised by means of exfoliation through chemical oxidation of graphite. The oxide form of graphene can easily be processed from solution. Deposition protocols for graphene derivatives in PSCs need to be performed in a way that incurs minimum damage to other underlying layers to preserve the integrity of the materials and prevent defect generation and promote scaling-up.¹¹⁴ The PSC fabrication methods involving graphene derivatives include electro-spraying,¹⁵² radio-frequency magnetron sputtering,¹²⁸ screen-printing,¹⁸⁵ lamination,¹¹⁹ doctor blade (blade coating)²¹ and spin-coating⁸² (Table 1). The spin-coating conditions (speed/time, annealing temperature, solvent) should be quantitatively managed to control morphology and thickness (thicker layers will be less transparent).^{17,24,54,156,186} Yang *et al.*⁵⁴ suggested that a perovskite thickness of at least 300 nm is sufficient for meaningful light absorption. The spin-coating method is the most popular and conventionally used in the fabrication of lab-scale devices.¹⁸⁷ However, large-scale manufacturing through spin-coating may produce pinholes, rough perovskite films and non-uniform coatings;^{30,87} hence, deposition studies still require more insights to avoid PCE losses during scale-up. Solution-coating processes support scaling-up, and typical methods include roll-to-roll,^{188,189} and thermal- and air-assisted blade coating at ambient conditions.¹¹³ Blade coating involves a solution/paste being dropped on a substrate followed by sweeping across the substrate with a blade-like implement.¹¹³ Graphene derivatives are compatible with scaling-up deposition methods as illustrated by a K-GO, ETL/perovskite interlayer, coated by means of a blade-slot die (area increase from 0.1 to 16 cm² slightly decreased PCE from 18.3% to 16.1%).¹⁸⁷ The roll-to-roll method has been successfully employed for PSCs with Li-GO as a TiO₂/perovskite interfacial material for a module with an area of 50.6 cm² and PCE of 12.6% (relative to 13.5% for an area of 0.01 cm²).³⁰ Screen-printing is another scaling-up technique in which the paste composition and conditions can tune properties.¹⁹⁰ The technique is suitable for scaling-up thin film deposition, usually at a speed greater than 6 m min⁻¹ but with long drying and sintering durations at high temperatures for better porosity.¹⁹¹ An investigation of the use of near-infrared heating was reported to substantially reduce the processing time at a speed of 2 m min⁻¹.¹⁹¹ The ink of graphene derivatives

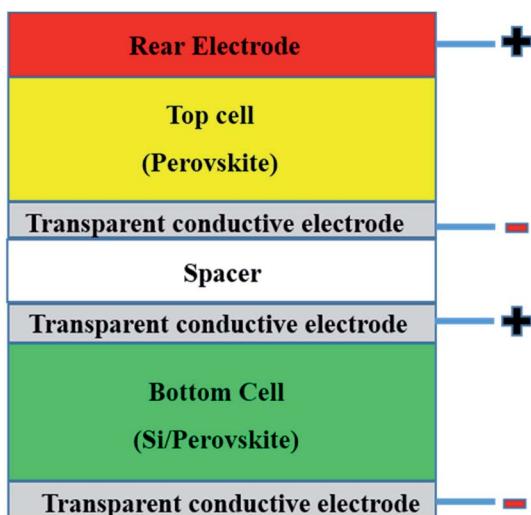


Fig. 4 Representative tandem structure involving PSCs in a four-terminal cell.



can be cheaply produced in high yields using non-toxic solvents to support integration into large PSC modules with competitive PCE.³⁰ Transfer of graphene derivatives through polymer carriers has led to performance deterioration; hence, it is difficult to fabricate effective TCEs.^{110,192} Despite knowledge gaps, graphene derivatives, particularly the oxide forms, are compatible with flexible substrates because of their high dispersion in solvents and the ease with which they can be coated; hence, future research directions can explore their use in improving the cost, stability, and performance of PSCs.

A fundamental aspect in PSC technology yet to be overcome is stability with respect to humidity (Table 1). Reference to the RH is necessary whenever investigations are done to ascertain the suitability of materials at different locations.⁷⁶ An analysis of PSC stability in terms of the decline in PCE upon graphene derivative inclusion as a p-i-n HTL component, solely a p-i-n HTL, the hole blocking layer component, active layer constituent, and ETL n-i-p constituent/blocking layer, after roughly 30 days show PCE declines of 12, 10, 50, 30 and 35%, respectively (Table 1). This may infer that graphene derivatives provide the greatest stability boost in p-i-n HTLs and the worst stability in p-i-n hole blocking layers. From the reviewed articles, the application of graphene derivatives in conventional PSCs has recorded the highest values of V_{oc} , J_{sc} and FF of >1 V, 26.96 mA cm⁻² and 82.80%, respectively (Table 1). Interestingly, the highest PCE boost of 21% was with graphene derivatives mixed with the perovskite layer.

5. Prospects and challenges

From the golden triangle (cost, PCE and stability) perspective,^{60,166} PSCs are promising for commercialisation due to both a fast-growing PCE and low fabrication costs. However, there is still a need to improve the stability and sustainability of devices, such as removing the use of toxic solvents in manufacturing⁴³ and mitigating the use of toxic lead through substitution or, failing this, sequestration and accountability throughout the lifecycle of devices. From the current review, it is clear that substantial studies have been dedicated to PSCs as an emerging technology, including optimisation of perovskite crystallisation, interfacial engineering and flexible substrate developments. The desired characteristics for PSC fabrication are homogeneous and continuous highly crystalline large perovskite grains, with high stability to moisture and atmospheric conditions. Since it may be challenging to develop a graphene derivative that addresses all shortfalls of PSCs and because some focused studies have shown their partial potential support to, particularly, performance and stability, future research should focus on the synergy between factors.

From the reviewed articles (Table 1), it is clear that the addition of graphene derivatives to PSC layers inhibits degradation of perovskite through defect passivation (promote uniform deposition to eliminate pinholes and promote perovskite growth), thus improving stability. This is because degradation is initiated at grain boundaries and defective regions.¹⁸⁷ Reduced defect intensity in perovskites and interface passivation induced by graphene derivatives also lessens charge traps;

therefore, promoting performance.^{60,95} Regulation of perovskite crystal growth by using graphene derivatives improves reproducibility of deposition of the active layer.¹⁶⁵ Graphene derivatives in PSC charge carriers (by virtue of demonstrated tuneable physicochemical properties *via* morphological and chemical transformations) promote performance by reducing electron/hole recombination.^{30,221,222} This is achieved through WF and band gap modulation that create appropriate energy level alignment between perovskite and charge carriers (ideal V_{oc}). Graphene derivatives also generate high J_{sc} and FF *via* interfacial chemical modifications that promote high electron extraction and transport from the ETL due to relatively higher conductivity. Additionally, graphene derivatives as dopants to the ETL suppress octahedral tilt in the perovskite monolayer to create nanoscale ferritic distortion with irreversible polarisation, thereby increasing electron extraction from the perovskite and minimising electron/hole recombination.³⁰

Future use of graphene derivatives in TCEs, though currently less common than pristine graphene, is motivated by the low-cost processing, controllable work function, and resistance to chemical degradation. The hydrophilicity of graphene derivatives has positive prospects for easy dispersions in polar solvents, making them applicable to low-temperature solution processing. RGO is a better option than pristine GO for PSC applications because it is associated with a smaller energy mismatch to the perovskite; hence, promoting electron extraction. The inclusion of graphene derivatives in PSCs needs critical optimisation since it may enhance other relevant parameters at the expense of PCE, as summarised below. For example, graphene oxide (GO and RGO) applications in PSCs still need a conductivity boost through doping or composite synthesis, development of practical, simplified large-scale synthesis, improved film transfer protocols, and optimisation insights towards thin and uniform materials. The film thicknesses of both perovskite and graphene derivatives are crucial factors that also influence the PCE. It is critical to note two contradicting phenomena discussed earlier; firstly, the need for high hydrophilicity during fabrication and sacrificial reaction centre/blocking purposes; secondly, the hydrophobicity requirement in the perovskite for long-term stability and functionality of PSCs. Also, on the one hand, GO-based composites have been reported in some studies to disrupt the homogeneity of spiro-OMeTAD, thereby causing a detrimental effect on PSCs by creating shunt pathways.¹²⁴ On the other hand, composites of GO, such as GO-Li, have also been reported to reduce moisture attack and enhance both stability and life-span through passivation of titania oxygen vacancies by acting as reactive centres and blocking channels to moisture attack.^{110,120} Hence, the current review suggests the need to monitor and balance the basic necessities, particularly cost and performance, of a functional PSC device.

Graphene derivatives are promising alternatives that can positively impact the costs involved in large-scale PSC manufacturing. Relevant current research gaps that involve the use of graphene derivatives in PSCs towards scalable, low temperature processed PSCs include:^{21,78}



(i) Improved mechanical infiltration of perovskite solution into graphene derivatives within short intervals and with better reproducibility.

(ii) Attainment of long-term stability under light exposure. Charge transport material should not cause photocatalytic degradation upon light exposure.

(iii) Increasing the temperature tolerance.

(iv) Enlargement of the active cell area.

Photo-induced decomposition is still challenging in PSCs because the perovskite/light interaction is strong; hence, future guidelines include using functionalised additives that also improve chemical interactions between ions.⁶³ Graphene derivatives have demonstrated, through an in-depth PSC degradation profile analysis under operational conditions (60% PCE retention (15.8%) relative to control PCE retention of 25% (13.6%) after 24 h light soaking),⁶¹ to improve long-term stability. This was attributed to a reduction in localised inhomogeneous light-induced decomposition of perovskite back to precursors, preventing I₂-induced m-TiO₂ degradation and inhibited diffusion of I₂ and Au across the whole device.

The use of graphene derivatives as PSC encapsulant materials is an underexplored area;¹⁰⁹ hence, as a future research direction, focus on in-depth degradation studies of typical devices is essential. Graphene-based materials are good candidates for PSC encapsulants due to their commendable optical properties.^{135,163,183} For instance, 60 µm of a GO encapsulant of a AgNW TCE achieved 83.8% transmittance at 550 nm (preserving 98.4% of the average transmittance of the pristine AgNW TCE).¹²² However, increasing the RGO loading can be detrimental as shown by Nouri *et al.*,²⁷ RGO loadings of 0, 0.05 and 0.8 wt% in the ETL achieved PCEs of 13.1%, 13.5% and 12.1%, respectively, due to reduced transparency. Hence, since charge carriers and encapsulants should permit sufficient light to reach the active layer, the future use of graphene derivatives in PSCs should critically investigate concentration dynamics. Also supporting this strategy, is the proven ability of small amounts of graphene derivatives in boosting PCE without altering optical properties.²²³ The realisation of suitable encapsulation materials and protocols prospectively supports the drive to achieve the approximated low 3.5 US cents per kW h target, which is competitive with current commercial solar cells and fossil energy resources upon scaling-up.¹⁶⁵

From the summarised past research, the attainability of the next stride of PSCs in commercial markets is highly possible if long-term stability, defect concentration minimisation, scalability and processability goals are met. This can be achieved through studies that provide a fundamental and thorough understanding of the effect of graphene derivatives on interfacial parameters and perovskite crystallisation. The modulation of the WF of graphene derivatives has also shown essential benefits in the functioning of PSCs. The current literature has presented a vast number of positive attributes linked with the inclusion of graphene derivatives in PSCs, but further in-depth studies and industrial up-scaling are required.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 D. Welsby, J. Price, S. Pye and P. Ekins, *Nature*, 2021, **597**, 230–234.
- 2 K. Vohra, E. Marais, L. J. Mickley, A. Vodonos and J. Schwartz, *Deaths from fossil fuel emissions higher than previously thought*, <https://news.harvard.edu/gazette/story/2021/02/deaths-from-fossil-fuel-emissions-higher-than-thought/>, accessed, 24 August, 2021.
- 3 F. Perera, *Int. J. Environ. Res. Public Health*, 2018, **15**, 16.
- 4 E. Mittelman, *Air pollution from fossil fuels costs \$8 billion per day, new research finds*, <https://e360.yale.edu/digest/air-pollution-from-fossil-fuels-costs-8-billion-per-day-new-research-finds>, accessed, 24 August 2021.
- 5 S. Maniarasu, T. B. Korukonda, V. Manjunath, E. Ramasamy, M. Ramesh and G. Veerappan, *Renewable Sustainable Energy Rev.*, 2018, **82**, 845–857.
- 6 B. Augustine, K. Remes, G. S. Lorite, J. Varghese and T. Fabritius, *Sol. Energy Mater. Sol. Cells*, 2019, **194**, 74–82.
- 7 M. Cai, Y. Wu, H. Chen, X. Yang, Y. Qiang and L. Han, *Adv. Sci.*, 2017, **4**, 1600269.
- 8 X. Tian, S. D. Stranks and F. You, *Nat. Sustain.*, 2021, **4**, 821–829.
- 9 M. L. Davies, *Joule*, 2020, **4**, 1626–1627.
- 10 Z. Song, C. L. McElvany, A. B. Phillips, I. Celik, P. W. Krantz, S. C. Watthage, G. K. Liyanage, D. Apul and M. J. Heben, *Energy Environ. Sci.*, 2017, **10**, 1297–1305.
- 11 P. Li, Z. Wu, H. Hu, Y. Zhang, T. Xiao, X. Lu, Z. Ren, G. Li, Z. Wu, J. Hao, H. L. Zhang and Z. Zheng, *ACS Appl. Mater. Interfaces*, 2020, **12**, 26050–26059.
- 12 P. Jiang, T. W. Jones, N. W. Duffy, K. F. Anderson, R. Bennett, M. Grigore, P. Marvig, Y. Xiong, T. Liu, Y. Sheng, L. Hong, X. Hou, M. Duan, Y. Hu, Y. Rong, G. J. Wilson and H. Han, *Carbon*, 2018, **129**, 830–836.
- 13 Y. Ding, B. He, J. Zhu, W. Zhang, G. Su, J. Duan, Y. Zhao, H. Chen and Q. Tang, *ACS Sustainable Chem. Eng.*, 2019, **7**, 19286–19294.



14 S. Zhang, Z. Liu, W. Zhang, Z. Jiang, W. Chen, R. Chen, Y. Huang, Z. Yang, Y. Zhang, L. Han and W. Chen, *Adv. Energy Mater.*, 2020, **10**, 2001610.

15 Q. Wali, F. J. Iftikhar, M. E. Khan, A. Ullah, Y. Iqbal and R. Jose, *Org. Electron.*, 2020, **78**, 105590.

16 Y. Xiong, X. Zhu, A. Mei, F. Qin, S. Liu, S. Zhang, Y. Jiang, Y. Zhou and H. Han, *Sol. RRL*, 2018, **2**, 1800002.

17 P. Zeng, W. Deng and M. Liu, *Sol. RRL*, 2020, **4**, 1900485.

18 B. Susrutha, L. Giribabu and S. P. Singh, *Chem. Commun.*, 2013, **58**, 4466–4483.

19 Y. Rong, Y. Hu, A. Mei, H. Tan, M. I. Saidaminov, S. I. Seok, M. D. McGehee, E. H. Sargent and H. Han, *Science*, 2018, **361**, eaat8235.

20 S. Bi, W. Zhao, Y. Sun, C. Jiang, Y. Liu, Z. He, Q. Li and J. Song, *Nanoscale Adv.*, 2021, **3**, 6659–6668.

21 L. Fagiolari and F. Bella, *Energy Environ. Sci.*, 2019, **12**, 3437–3472.

22 M. Batmunkh, C. J. Shearer, M. J. Biggs and J. G. Shapter, *J. Mater. Chem. A*, 2016, **4**, 2605–2616.

23 S. S. Mali, C. S. Shim and C. K. Hong, *Sci. Rep.*, 2015, **5**, 11424.

24 Q.-D. Yang, J. Li, Y. Cheng, H.-W. Li, Z. Guan, B. Yub and S.-W. Tsang, *J. Mater. Chem. A*, 2017, **5**, 9852–9858.

25 Z. Wu, S. Bai, J. Xiang, Z. Yuan, Y. Yang, W. Cui, X. Gao, Z. Liu, Y. Jin and B. Sun, *Nanoscale*, 2014, **6**, 10505–10510.

26 G. S. Han, Y. H. Song, Y. U. Jin, J.-W. Lee, N.-G. Park, B. K. Kang, J.-K. Lee, I. S. Cho, D. H. Yoon and H. S. Jung, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23521–23526.

27 E. Nouri, M. R. Mohammadi, Z.-X. Xu, V. Dracopoulos and P. Lianos, *Phys. Chem. Chem. Phys.*, 2018, **20**, 2388–2395.

28 Q. Wang, N. Phung, D. D. Girolamo, P. Vivo and A. Abate, *Energy Environ. Sci.*, 2019, **12**, 865–886.

29 S. Ameen, M. S. Akhtar, M. Nazim, E.-B. Kim, M. K. Nazeeruddin and H.-S. Shin, *Electrochim. Acta*, 2019, **319**, 885–894.

30 A. Agresti, S. Pescetelli, A. L. Palma, A. E. D. R. Castillo, D. Konios, G. Kakavelakis, S. Razza, L. Cinà, E. Kymakis, F. Bonaccorso, A. D. Carlo, D. Konios, G. Kakavelakis, S. Razza, L. Cinà, E. Kymakis, F. Bonaccorso and A. D. Carlo, *ACS Energy Lett.*, 2017, **2**, 279–287.

31 H.-S. Kim, C.-R. Lee, J.-H. Im, K.-B. Lee, T. Moehl, A. Marchioro, S.-J. Moon, R. Humphry-Baker, J.-H. Yum, J. E. Moser, M. Grätzel and N.-G. Park, *Sci. Rep.*, 2012, **2**, 591.

32 *Best Research-Cell Efficiencies Chart*, <https://www.nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.20200218.pdf>, accessed, 20 August 2021.

33 M. A. Green, E. D. Dunlop, J. Hohl-Ebinger, M. Yoshita, N. Kopidakis and X. Hao, *Prog. Photovoltaics*, 2021, **29**, 657–667.

34 T. Umeyama, D. Matano, J. Baek, S. Gupta, S. Ito, V. R. Subramanian and H. Imahori, *Chem. Lett.*, 2015, **44**, 1410–1412.

35 J. A. Hong, E. D. Jung, J. C. Yu, D. W. Kim, Y. S. Nam, I. Oh, E. Lee, J.-W. Yoo, S. Cho and M. H. Song, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2417–2423.

36 D. Li, J. Cui, H. Li, D. Huang, M. Wang and Y. Shen, *Sol. Energy*, 2016, **131**, 176–182.

37 J. T.-W. Wang, J. M. Ball, E. M. Barea, A. Abate, J. A. Alexander-Webber, J. Huang, M. Saliba, I. n. Mora-Sero, J. Bisquert, H. J. Snaith and R. J. Nicholas, *Nano Lett.*, 2014, **14**, 724–730.

38 J. He, C.-F. Ng, K. Y. Wong, W. Liu and T. Chen, *ChemPlusChem*, 2016, **81**, 1292–1298.

39 D. Lasrado, S. Ahankari and K. Kar, *J. Appl. Polym. Sci.*, 2020, **137**, 48959.

40 H. S. Jung, G. S. Han, N.-G. Park and M. J. Ko, *Joule*, 2019, **3**, 1850–1880.

41 E. D. Jung, A. K. Harit, D. H. Kim, C. H. Jang, J. H. Park, S. Cho, M. H. Song and H. Y. Woo, *Adv. Mater.*, 2020, **32**, 2002333.

42 E. Muchuweni, B. S. Martincigh and V. O. Nyamori, *Adv. Energy Sustainability Res.*, 2021, **2**, 2100050.

43 A. J. Doolin, R. G. Charles, C. S. P. De Castro, R. G. Rodriguez, E. V. Péan, R. Patidar, T. Dunlop, C. Charbonneau, T. Watson and M. L. Davies, *Green Chem.*, 2021, **23**, 2471–2486.

44 L. Yang, Y. Dall'Agnese, K. Hantanasisirakul, C. E. Shuck, K. Maleski, M. Alhabeb, G. Chen, Y. Gao, Y. Sanehira, A. K. Jena, L. Shen, C. Dall'Agnese, X.-F. Wang, Y. Gogotsi and T. Miyasaka, *J. Mater. Chem. A*, 2019, **7**, 5635–5642.

45 J. Ren, Q. Luo, Q. Hou, H. Chen, T. Liu, H. He, J. Wang, Q. Shao, M. Dong, S. Wu, N. Wang, J. Lin and Z. Guo, *ChemElectroChem*, 2019, **6**, 3167–3174.

46 H.-S. Kim, B. Yang, M. M. Stylianakis, E. Kymakis, S. M. Zakeeruddin, M. Grätzel and A. Hagfeldt, *Cell Rep. Phys. Sci.*, 2020, **1**, 100053.

47 Y. Yang, J. Xiao, H. Wei, L. Zhu, D. Li, Y. Luo, H. Wu and Q. Meng, *RSC Adv.*, 2014, **4**, 52825–52830.

48 S. M. P. Meroni, K. E. A. Hooper, T. Dunlop, J. A. Baker, D. Worsley, C. Charbonneau and T. M. Watson, *Energies*, 2020, **13**, 1589.

49 W.-C. Oh, K. Y. Cho, C.-H. Jung and Y. Areerob, *J. Photochem. Photobiol. A*, 2019, **372**, 11–20.

50 Y. Duan, G. You, K. Sun, Z. Zhu, X. Liao, L. Lv, H. Tang, B. Xu and L. He, *Nanoscale Adv.*, 2021, **3**, 6271–6293.

51 J.-S. Yeo, C.-H. Lee, D. Jang, S. Lee, S. M. Jo, H.-I. Joh and D.-Y. Kim, *Nano Energy*, 2016, **30**, 667–676.

52 J. A. Hong, E. D. Jung, J. C. Yu, D. W. Kim, Y. S. Nam, I. Oh, E. Lee, J. W. Yoo, S. Cho and M. H. Song, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2417–2423.

53 B. Dou, J. B. Whitaker, K. Bruening, D. T. Moore, L. M. Wheeler, J. Ryter, N. J. Breslin, J. J. Berry, S. M. Garner, F. Barnes, S. E. Shaheen, C. J. Tassone, K. Zhu and M. F. A. M. v. Hest, *ACS Energy Lett.*, 2018, **3**, 2558–2565.

54 D. Yang, R. Yang, S. Priya and S. F. Liu, *Angew. Chem., Int. Ed.*, 2019, **58**, 4466–4483.

55 C. Worsley, D. Raptis, S. Meroni, A. Doolin, R. Garcia-Rodriguez, M. Davies and T. Watson, *Energy Technol.*, 2021, **9**, 2100312.

56 D. Y. Lee, S. I. Na and S. S. Kim, *Nanoscale*, 2016, **8**, 1513–1522.

57 R. Singh, P. K. Singh, B. Bhattacharya and H.-W. Rhee, *Appl. Mater. Today*, 2019, **14**, 175–200.



58 E. W. Jones, P. J. Holliman, A. Connell, M. L. Davies, J. Baker, R. J. Hobbs, S. Ghosh, L. Furnell, R. Anthony and C. Pleydell-Pearce, *Chem. Commun.*, 2016, **52**, 4301–4304.

59 F. Gao, Y. Zhao, X. Zhang and J. You, *Adv. Energy Mater.*, 2020, **10**, 1902650.

60 L. Meng, J. You and Y. Yang, *Nat. Commun.*, 2018, **9**, 5265.

61 Y. Busby, A. Agresti, S. Pescetelli, A. Di Carlo, C. Noel, J.-J. Pireaux and L. Houssiau, *Mater. Today Energy*, 2018, **9**, 1–10.

62 E. V. Péan, C. S. De Castro, S. Dimitrov, F. De Rossi, S. Meroni, J. Baker, T. Watson and M. L. Davies, *Adv. Funct. Mater.*, 2020, **30**, 1909839.

63 N. Li, Y. Luo, Z. Chen, X. Niu, X. Zhang, J. Lu, R. Kumar, J. Jiang, H. Liu, X. Guo, B. Lai, G. Brocks, Q. Chen, S. Tao, D. P. Fenning and H. Zhou, *Joule*, 2020, **4**, 1743–1758.

64 B. Chen, J. Song, X. Dai, Y. Liu, P. N. Rudd, X. Hong and J. Huang, *Adv. Mater.*, 2019, **31**, 1902413.

65 H.-Q. Wang, S. Wang, L. Chen, Z. Yin, S. Mei, Y. Zhong, Y. Yao, N. Li, J. Wang and W. Song, *Sol. Energy Mater. Sol. Cells*, 2021, **230**, 111278.

66 Y. Liu, C. Xie, W. Tan, X. Liu, Y. Yuan, Q. Xie, Y. Li and Y. Gao, *Org. Electron.*, 2019, **71**, 123–130.

67 J. Burschka, N. Pellet, S.-J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin and M. Grätzel, *Nature*, 2013, **499**, 316–319.

68 X. Zhang, X. Chen, Y. Chen, N. A. N. Ouedraogo, J. Li, X. Bao, C. B. Han, Y. Shirai, Y. Zhang and H. Yan, *Nanoscale Adv.*, 2021, **3**, 6128–6137.

69 K. Domanski, E. A. Alharbi, A. Hagfeldt, M. Grätzel and W. Tress, *Nat. Energy*, 2018, **3**, 61–67.

70 F. Lang, O. Shargaieva, V. V. Brus, H. C. Neitzert, J. Rappich and N. H. Nickel, *Adv. Mater.*, 2018, **30**, 1702905.

71 S.-W. Lee, S. Kim, S. Bae, K. Cho, T. Chung, L. E. Mundt, S. Lee, S. Park, H. Park, M. C. Schubert, S. W. Glunz, Y. Ko, Y. Jun, Y. Kang, H.-S. Lee and D. Kim, *Sci. Rep.*, 2016, **6**, 38150.

72 E. V. Péan, C. S. De Castro and M. L. Davies, *Mater. Lett.*, 2019, **243**, 191–194.

73 J. Wei, Q. Wang, J. Huo, F. Gao, Z. Gan, Q. Zhao and H. Li, *Adv. Energy Mater.*, 2021, **11**, 2002326.

74 F. Fu, S. Pisoni, Q. Jeangros, J. Sastre-Pellicer, M. Kawecki, A. Paracchino, T. Moser, J. Werner, C. Andres, L. Duchêne, P. Fiala, M. Rawlence, S. Nicolay, C. Ballif, A. N. Tiwari and S. Buecheler, *Energy Environ. Sci.*, 2019, **12**, 3074–3088.

75 B. Rivkin, P. Fassl, Q. Sun, A. D. Taylor, Z. Chen and Y. Vaynzof, *ACS Omega*, 2018, **3**, 10042–10047.

76 B. Martín-García, Y. Bi, M. Prato, D. Spirito, R. Krahne, G. Konstantatos and I. Moreels, *Sol. Energy Mater. Sol. Cells*, 2018, **183**, 1–7.

77 P. S. Chandrasekhara, A. Dubey and Q. Qiao, *Sol. Energy*, 2020, **197**, 78–83.

78 V. Ferguson, S. R. P. Silva and W. Zhang, *Energy Environ. Mater.*, 2019, **2**, 107–118.

79 K. Petridis, G. Kakavelakis, M. M. Stylianakis and E. Kymakis, *Chem.-Asian J.*, 2018, **13**, 240–249.

80 C.-C. Chung, S. Narra, E. Jokar, H.-P. Wu and E. W.-G. Diau, *J. Mater. Chem. A*, 2017, **5**, 13957–13965.

81 E. Kymakis and D. Konios, in *The Future of Semiconductor Oxides in Next-Generation Solar Cells*, Elsevier, 2018, vol. 9, pp. 357–394.

82 Q. Liu, C. Leng and J. Yuan, *Presented in Part at the IOP Conference Series: Materials Science and Engineering*, 2018.

83 J. Cao, Y.-M. Liu, X. Jing, J. Yin, J. Li, B. Xu, Y.-Z. Tan and N. Zheng, *J. Am. Chem. Soc.*, 2015, **137**, 10914–10917.

84 M. Zhu, W. Liu, W. Ke, L. Xie, P. Dong and F. Hao, *ACS Appl. Mater. Interfaces*, 2019, **11**, 666–673.

85 S. Feng, Y. Yang, M. Li, J. Wang, Z. Cheng, J. Li, G. Ji, G. Yin, F. Song, Z. Wang, J. Li and X. Gao, *ACS Appl. Mater. Interfaces*, 2016, **8**, 14503–14512.

86 J. Yang, S. Xiong, T. Qu, Y. Zhang, X. He, X. Guo, Q. Zhao, S. Braun, J. Chen, J. Xu, Y. Li, X. Liu, C. Duan, J. Tang, M. Fahlman and Q. Bao, *ACS Appl. Mater. Interfaces*, 2019, **11**, 13491–13498.

87 J.-P. Correa-Baena, M. Saliba, T. Buonassisi, M. Grätzel, A. Abate, W. Tress and A. Hagfeldt, *Science*, 2017, **358**, 739–744.

88 S.-m. Ahn, E. D. Jung, S.-H. Kim, H. Kim, S. Lee, M. H. Song and J.-Y. Kim, *Nano Lett.*, 2019, **19**, 3707–3715.

89 F. Zhang and K. Zhu, *Adv. Energy Mater.*, 2020, **10**, 1902579.

90 Q. Fu, X. Tang, B. Huang, T. Hu, L. Tan, L. Chen and Y. Chen, *Adv. Sci.*, 2018, **5**, 1700387.

91 Q. Zhao, G. R. Li, J. Song, Y. Zhao, Y. Qiang and X. P. Gao, *Sci. Rep.*, 2016, **6**, 38670.

92 H.-Y. Chu, J.-Y. Hong, C.-F. Huang, J.-Y. Wu, T.-L. Wang, T.-M. Wu and R.-H. Lee, *Cellulose*, 2019, **26**, 9229–9239.

93 Q. A. Yousif and S. Agbolaghi, *Macromol. Res.*, 2020, **28**, 425–432.

94 S. Mabrouk, B. Bahrami, H. Elbohy, K. M. Reza, A. Gurung, M. Liang, F. Wu, M. Wang, S. Yang and Q. Qiao, *InfoMat*, 2020, **2**, 928–941.

95 J. V. Milić, N. Arora, M. I. Dar, S. M. Zakeeruddin and M. Grätzel, *Adv. Mater. Interfaces*, 2018, **5**, 1800416.

96 E. T. Mombeshora and A. Stark, *Biomass Convers. Biorefin.*, 2021, DOI: 10.1007/s13399-021-01499-6.

97 M. Acik and S. B. Darling, *J. Mater. Chem. A*, 2016, **4**, 6185–6235.

98 E. T. Mombeshora and A. Stark, *Mater. Chem. Phys.*, 2022, **277**, 125535.

99 T. Liu, Z. Wang, L. Lou, S. Xiao, S. Zheng and S. Yang, *Sol. RRL*, 2020, **4**, 1900278.

100 M. Duan, Y. Rong, A. Mei, Y. Hu, Y. Sheng, Y. Guan and H. Han, *Carbon*, 2017, **120**, 71–76.

101 H. Luo, X. Lin, X. Hou, L. Pan, S. Huang and X. Chen, *Nano-Micro Lett.*, 2017, **9**, 39.

102 T.-H. Han, J.-W. Lee, C. Choi, S. Tan, C. Lee, Y. Zhao, Z. Dai, N. D. Marco, S.-J. Lee, S.-H. Bae, Y. Yuan, H. M. Lee, Y. Huang and Y. Yang, *Nat. Commun.*, 2019, **10**, 520.

103 T. Mahmoudi, Y. Wang and Y.-B. Hahn, *ACS Energy Lett.*, 2019, **4**, 235–241.

104 Y. Numata, Y. Sanehira and T. Miyasaka, *ACS Appl. Mater. Interfaces*, 2016, **8**, 4608–4615.



105 Y. Wang, T. Mahmoudi and Y.-B. Hahn, *Adv. Energy Mater.*, 2020, **10**, 2000967.

106 M. Li, W.-W. Zuo, Q. Wang, K.-L. Wang, M.-P. Zhuo, H. Köbler, C. E. Halbig, S. Eigler, Y.-G. Yang, X.-Y. Gao, Z.-K. Wang, Y. Li and A. Abate, *Adv. Energy Mater.*, 2020, **10**, 1902653.

107 N. Balis, A. A. Zaky, C. Athanasekou, A. M. T. Silva, E. Sakellis, M. Vasilopoulou, T. Stergiopoulos, A. G. Kontos and P. Falaras, *J. Photochem. Photobiol. A*, 2020, **386**, 112141.

108 J. Niu, D. Yang, X. Ren, Z. Yang, Y. Liu, X. Zhu, W. Zhao and S. F. Liu, *Org. Electron.*, 2017, **48**, 165–171.

109 J. Zhang, J. Fan, B. Cheng, J. Yu and W. Ho, *Sol. RRL*, 2020, **4**, 2000502.

110 H. Su, T. Wu, D. Cui, X. Lin, X. Luo, Y. Wang and L. Han, *Small Methods*, 2020, **4**, 2000507.

111 T. Mahmoudi, Y. Wang and Y.-B. Hahn, *Nano Energy*, 2021, **79**, 105452.

112 M. Hadadian, J.-H. Smått and J.-P. Correa-Baena, *Energy Environ. Sci.*, 2020, **13**, 1377–1407.

113 D. Bogachuk, S. Zouhair, K. Wojciechowski, B. Yang, V. Babu, L. Wagner, B. Xu, J. Lim, S. Mastroianni, H. Pettersson, A. Hagfeldt and A. Hinsch, *Energy Environ. Sci.*, 2020, **13**, 3880–3916.

114 J.-J. Park, Y.-J. Heo, J.-M. Yun, Y. Kim, S. C. Yoon, S.-H. Lee and D.-Y. Kim, *ACS Appl. Mater. Interfaces*, 2020, **12**, 42811–42820.

115 E. Nouri, M. R. Mohammadi and P. Lianos, *Chem. Commun.*, 2017, **53**, 1630–1633.

116 T. Liu, D. Kim, H. Han, A. R. b. M. Yusoff and J. Jang, *Nanoscale*, 2015, **7**, 10708–10718.

117 D. S. Mann, Y.-H. Seo, S.-N. Kwon and S.-I. Na, *J. Alloys Compd.*, 2020, **812**, 152091.

118 U. Kim, K. Choi, K. H. Park, J. Lee, Y. Choi, J. Seo, N. K. Oh, S. Jung, H. Yang, J. H. Lee, C. Yang and H. Park, *ChemNanoMat*, 2019, **5**, 776–783.

119 H. Sung, N. Ahn, M. S. Jang, J.-K. Lee, H. Yoon, N.-G. Park and M. Choi, *Adv. Energy Mater.*, 2016, **6**, 1501873.

120 A. Agresti, S. Pescetelli, L. Cinà, D. Konios, G. Kakavelakis, E. Kymakis and A. D. Carlo, *Adv. Funct. Mater.*, 2016, **26**, 2686–2694.

121 E. T. Mombeshora, P. G. Ndungu and V. O. Nyamori, *Electrochim. Acta*, 2017, **258**, 467–476.

122 H. Chen, M. Li, X. Wen, Y. Yang, D. He, W. C. H. Choy and H. Lu, *Nanomaterials*, 2019, **9**, 193.

123 A. Giuri, S. Masi, S. Colella, A. Listorti, A. Rizzo, G. Gigli, A. Liscio, E. Treossi, V. Palermo, S. Rella, C. Malitesta and C. E. Corcione, *IEEE Trans. Nanotechnol.*, 2016, **15**, 725–730.

124 K. T. Cho, G. Grancini, Y. Lee, D. Konios, S. Paek, E. Kymakis and M. K. Nazeeruddin, *ChemSusChem*, 2016, **9**, 3040–3044.

125 N. Zibouche, G. Volonakis and F. Giustino, *J. Phys. Chem. C*, 2018, **122**, 16715–16726.

126 J. A. Hong, E. D. Jung, J. C. Yu, D. W. Kim, Y. S. Nam, I. Oh, E. Lee, J.-W. Yoo, S. Cho and M. H. Song, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2417–2423.

127 J. Bouclé and N. Herlin-Boime, *Synth. Met.*, 2016, **222**, 3–16.

128 L.-C. Chen, J.-C. Lin, K.-L. Lee and Z.-L. Tseng, *Presented in Part at the 7th IEEE International Symposium on Next-Generation Electronics (ISNE 2018)*, 2018.

129 V. K. Ravi, B. Mondal, V. V. Nawale and A. Nag, *ACS Omega*, 2020, **5**, 29631–29641.

130 G. Phipps, C. Mikolajczak and T. Guckes, *Renew. Energy Focus*, 2008, **9**, 56–59.

131 S. Zandi, P. Saxena and N. E. Gorji, *Sol. Energy*, 2020, **197**, 105–110.

132 A. K. Kang, M. H. Zandi and N. E. Gorji, *Opt. Quantum Electron.*, 2019, **51**, 91.

133 H. Su, J. Xiao, Q. Li, C. Peng, X. Zhang, C. Mao, Q. Yao, Y. Lu, Z. Ku, J. Zhong, W. Li, Y. Peng, F. Huang and Y.-b. Cheng, *Mater. Sci. Semicond. Process.*, 2020, **107**, 104809.

134 Y. Ahn, H. Oh, Y. Yoon, W. K. Park, W. S. Yang and J.-W. Kang, *J. Environ. Chem. Eng.*, 2017, **5**, 3882–3894.

135 S. Kim, S. H. Shin and S.-H. Choi, *J. Alloys Compd.*, 2019, **786**, 614–620.

136 T. Iqbal, M. Haqnawaz, M. Sultan, M. B. Tahir, M. I. Khan, K. N. Riaz, M. Ijaz and M. Rafique, *Int. J. Energy Res.*, 2018, **42**, 4866–4874.

137 J. H. Heo, D. H. Shin, M. H. Jang, M. L. Lee, M. G. Kang and S. H. Im, *J. Mater. Chem. A*, 2017, **5**, 21146–21152.

138 M. Hadadian, J.-P. Correa-Baena, E. K. Goharshadi, A. Ummadisingu, J.-Y. Seo, J. Luo, S. Gholipour, S. M. Zakeeruddin, M. Saliba, A. Abate, M. Grätzel and A. Hagfeldt, *Adv. Mater.*, 2016, **28**, 8681–8686.

139 C. Redondo-Obispo, T. S. Ripolles, S. Cortijo-Campos, A. L. Álvarez, E. Climent-Pascual, A. d. Andrés and C. Coy, *Mater. Des.*, 2020, **191**, 108587.

140 R. Hu, L. Chu, J. Zhang, X. a. Li and W. Huang, *J. Power Sources*, 2017, **361**, 259–275.

141 X. Hu, X. Meng, L. Zhang, Y. Zhang, Z. Cai, Z. Huang, M. Su, Y. Wang, M. Li, F. Li, X. Yao, F. Wang, W. Ma, Y. Chen and Y. Song, *Joule*, 2019, **3**, 2205–2218.

142 W. Li, H. Dong, X. Guo, N. Li, J. Li, G. Niu and L. Wang, *J. Mater. Chem. A*, 2014, **2**, 20105–20111.

143 T. P. Nguyen, D. L. T. Nguyen, V.-H. Nguyen, T.-H. Le, D.-V. N. Vo, Q. V. Ly, S. Y. Kim and Q. V. Le, *Polymers*, 2019, **11**, 1858.

144 Ç. Şahin, H. Diker, D. Sygkridou, C. Varlikli and E. Stathatos, *Renewable Energy*, 2020, **146**, 1659–1666.

145 F. Wang, M. Endo, S. Mouri, Y. Miyauchi, Y. Ohno, A. Wakamiya, Y. Murata and K. Matsuda, *Nanoscale*, 2016, **8**, 11882–11888.

146 N. E. Safie, M. A. Azam, M. F. A. Aziz and M. Ismail, *Int. J. Energy Res.*, 2021, **45**, 1347–1374.

147 C. Gao, S. Yuan, K. Cui, Z. Qiu, S. Ge, B. Cao and J. Yu, *Sol. RRL*, 2018, **2**, 1800175.

148 J.-W. Lee, T.-Y. Lee, P. J. Yoo, M. Grätzel, S. Mhaisalkar and N.-G. Park, *J. Mater. Chem. A*, 2014, **2**, 9251–9259.

149 P. Liu, Z. Yu, N. Cheng, C. Wang, Y. Gong, S. Bai and X.-Z. Zhao, *Electrochim. Acta*, 2016, **213**, 83–88.

150 L. Zhang, T. Liu, L. Liu, M. Hu, Y. Yang, A. Mei and H. Han, *J. Mater. Chem. A*, 2015, **3**, 9165–9170.



151 R. Dileep, G. Kesavan, V. Reddy, M. K. Rajbhar, S. Shanmugasundaram, E. Ramasamy and G. Veerappan, *Sol. Energy*, 2019, **187**, 261–268.

152 H. Li, L. Tao, F. Huang, Q. Sun, X. Zhao, J. Han, Y. Shen and M. Wang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 38967–38976.

153 S. S. Mali, C. S. Shim, H. Kim and C. K. Hong, *J. Mater. Chem. A*, 2016, **4**, 12158–12169.

154 Q. Luo, Y. Zhang, C. Liu, J. Li, N. Wang and H. Lin, *J. Mater. Chem. A*, 2015, **3**, 15996–16004.

155 M. Dadashbeik, D. Fathi and M. Eskandari, *Sol. Energy*, 2020, **207**, 917–924.

156 J.-H. Im, H.-S. Kim and N.-G. Park, *APL Mater.*, 2014, **2**, 1–8.

157 H.-S. Ko, J.-W. Lee and N.-G. Park, *J. Mater. Chem. A*, 2015, **3**, 808–8815.

158 Y. Wang, Y. Hu, D. Han, Q. Yuan, T. Cao, N. Chen, D. Zhou, H. Cong and L. Feng, *Org. Electron.*, 2019, **70**, 63–70.

159 D. Selvakumar, G. Murugadoss, A. Alsalme, A. M. Alkathiri and R. Jayavel, *Sol. Energy*, 2018, **163**, 564–569.

160 M. Habib, M. Feteha, M. Soliman, A. A. Motagaly, S. El-Sheikh and S. Ebrahim, *J. Mater. Sci.: Mater. Electron.*, 2020, **31**, 18870–18882.

161 B. Park, S.-M. Kang, G.-W. Lee, C. H. Kwak, M. Rethinasabapathy and Y. S. Huh, *Ind. Eng. Chem. Res.*, 2020, **59**, 793–801.

162 J. S. Cho, W. Jang, S. C. Mun, M. Yi, J. H. Park and D. H. Wang, *Carbon*, 2018, **139**, 564–571.

163 A. Yi, S. Chae, S. Won, H.-J. Jung, I. H. Cho, J.-H. Kim and H. J. Kim, *Nano Energy*, 2020, **77**, 105182.

164 N. F. Ramli, P. N. A. Fahsyar, N. A. Ludin, M. A. M. Teridi, M. A. Ibrahim and S. Sepeai, *Mater. Chem. Phys.*, 2021, **257**, 123798.

165 P. Mariani, L. Najafi, G. Bianca, M. I. Zappia, L. Gabatell, A. Agresti, S. Pescetelli, A. Di Carlo, S. Bellani and F. Bonaccorso, *ACS Appl. Mater. Interfaces*, 2021, **13**, 22368–22380.

166 K. L. Wang, Y. H. Zhou, Y. H. Lou and Z. K. Wang, *Chem. Sci.*, 2021, **12**, 11936–11954.

167 K. Yuwawech, J. Woothikanokkhan, S. Wanwong and S. Tanpichai, *J. Appl. Polym. Sci.*, 2017, **134**, 45010.

168 E. Nouri, M. R. Mohammadi and P. Lianos, *Carbon*, 2018, **126**, 208–214.

169 A. G. Boldyreva, I. S. Zhidkov, S. Tsarev, A. F. Akbulatov, M. M. Tepliakova, Y. S. Fedotov, S. I. Bredikhin, E. Y. Postnova, S. Y. Luchkin, E. Z. Kurmaev, K. J. Stevenson and P. A. Troshin, *ACS Appl. Mater. Interfaces*, 2020, **12**, 19161–19173.

170 T. Salim, S. Sun, Y. Abe, A. Krishna, A. C. Grimsdale and Y. M. Lam, *J. Mater. Chem. A*, 2015, **3**, 8943–8969.

171 K. Yan, Z. Wei, J. Li, H. Chen, Y. Yi, X. Zheng, X. Long, Z. Wang, J. Wang, J. Xu and S. Yang, *Small*, 2015, **11**, 2269–2274.

172 L. Huang, Z. Hu, J. Xu, X. Sun, Y. Du, J. Ni, H. Cai, J. Li and J. Zhang, *Sol. Energy Mater. Sol. Cells*, 2016, **152**, 118–124.

173 X. Dai, P. Koshy, C. C. Sorrell, J. Lim and J. S. Yun, *Energies*, 2020, **13**, 6335.

174 Y. Li, J. Zhu, Y. Huang, F. Liu, M. Lv, S. Chen, L. Hu, J. Tang, J. Yao and S. Dai, *RSC Adv.*, 2015, **5**, 28424–28429.

175 S. S. Shin, W. S. Yang, J. H. Noh, J. H. Suk, N. J. Jeon, J. H. Park, J. S. Kim, W. M. Seong and S. I. Seok, *Nat. Commun.*, 2015, **6**, 7410.

176 X. Ma, Q. Deng, L. Wang, X. Zheng, S. Wang, Q. Wang, L. Chen, L. Huang, X. Ouyang and S. Cao, *RSC Adv.*, 2019, **9**, 9348–9353.

177 B. Xie, Y. Zhang, Y. Li, W. Chen, X. Hu and S. Zhang, *J. Mater. Sci.: Mater. Electron.*, 2020, **31**, 6248–6254.

178 M. M. Tavakoli, R. Tavakoli, Z. Nourbakhsh, A. Waleed, U. S. Virk and Z. Fan, *Adv. Mater. Interfaces*, 2016, **3**, 1500790.

179 Z. Zhang, Z. Li, L. Meng, S. Y. Lien and P. Gao, *Adv. Funct. Mater.*, 2020, **30**, 2001904.

180 A. B. Djurišić, F. Z. Liu, H. W. Tam, M. K. Wong, A. Ng, C. Surya, W. Chen and Z. B. He, *Prog. Quantum Electron.*, 2017, **53**, 1–37.

181 R. K. Kothandaraman, Y. Jiang, T. Feurer, A. N. Tiwari and F. Fu, *Small Methods*, 2020, **4**, 2000395.

182 F. Lang, M. A. Gluba, S. Albrecht, J. Rappich, L. Korte, B. Rech and N. H. Nickel, *J. Phys. Chem. Lett.*, 2015, **6**, 2745–2750.

183 J. Zhou, Z. Ren, S. Li, Z. Liang, C. Surya and H. Shen, *Mater. Lett.*, 2018, **220**, 82–85.

184 C. Ciminelli, F. Dell'Olio, G. Brunetti, D. Conteduca and M. N. Armenise, *Presented in Part at the 20th International Conference on Transparent Optical Networks 2018*, Bucharest, Romania, 1–5 July 2018.

185 D. Pourjafari, S. M. P. Meroni, D. Peralta Domínguez, R. Escalante, J. Baker, A. S. Monroy, A. Walters, T. Watson and G. Oskam, *Energies*, 2022, **15**, 641.

186 M. Li, X. Xu, Y. Xie, H.-W. Li, Y. Ma, Y. Cheng and S.-W. Tsang, *J. Mater. Chem. A*, 2019, **7**, 9578–9586.

187 L. A. Castriotta, F. Matteocci, L. Vesce, L. Cina, A. Agresti, S. Pescetelli, A. Ronconi, M. Loffler, M. M. Stylianakis, F. Di Giacomo, P. Mariani, M. Stefanelli, E. M. Speller, A. Alfano, B. Paci, A. Generosi, F. Di Fonzo, A. Petrozza, B. Rellinghaus, E. Kymakis and A. Di Carlo, *ACS Appl. Mater. Interfaces*, 2021, **13**, 11741–11754.

188 D. Burkitt, R. Patidar, P. Greenwood, K. Hooper, J. McGettrick, S. Dimitrov, M. Colombo, V. Stoichkov, D. Richards, D. Beynon, M. Davies and T. Watson, *Sustainable Energy Fuels*, 2020, **4**, 3340–3351.

189 R. Swartwout, R. Patidar, E. Belliveau, B. Dou, D. Beynon, P. Greenwood, N. Moody, D. deQuilettes, M. Bawendi, T. Watson and V. Bulovic, *Sol. RRL*, 2022, **6**, 2100567.

190 M. Xu, W. Ji, Y. Sheng, Y. Wu, H. Cheng, J. Meng, Z. Yan, J. Xu, A. Mei, Y. Hu, Y. Rong and H. Han, *Nano Energy*, 2020, **74**, 104842.

191 J. Baker, K. Hooper, S. Meroni, A. Pockett, J. McGettrick, Z. Wei, R. Escalante, G. Oskam, M. Carnie and T. Watson, *J. Mater. Chem. A*, 2017, **5**, 18643–18650.

192 V.-D. Tran, S. V. N. Pammi, B.-J. Park, Y. Han, C. Jeon and S.-G. Yoon, *Nano Energy*, 2019, **65**, 104018.

193 K. Ahmad and S. M. Mobin, *New J. Chem.*, 2017, **41**, 14253–14258.

194 S. Wang, X. Huang, H. Sun and C. Wu, *Nanoscale Res. Lett.*, 2017, **12**, 619.



195 X. Huang, H. Guo, J. Yang, K. Wang, X. Niu and X. Liu, *Org. Electron.*, 2016, **39**, 288–295.

196 H. Guo, X. Huang, B. Pu, J. Yang, H. Chen, Y. Zhou, J. Yang, Y. Li, Z. Wang and X. Niu, *RSC Adv.*, 2017, **7**, 50410–50419.

197 Y. Wang, S. Wang, X. Chen, Z. Li, J. Wang, T. Li and X. Deng, *J. Mater. Chem. A*, 2018, **6**, 4860–4867.

198 G. Kakavelakis, T. Maksudov, D. Konios, I. Paradisanos, G. Kioseoglou, E. Stratakis and E. Kymakis, *Adv. Energy Mater.*, 2017, **7**, 1602120.

199 D. Benetti, E. Jokar, C.-H. Yu, A. Fathi, H. Zhao, A. Vomiero, E. W.-G. Diau and F. Rosei, *Nano Energy*, 2019, **62**, 781–790.

200 J. Kim, M. A. M. Teridi, A. R. b. M. Yusoff and J. Jang, *Sci. Rep.*, 2016, **6**, 27773.

201 E. Jokar, Z. Y. Huang, S. Narra, C.-Y. Wang, V. Kattoor, C.-C. Chung and E. W.-G. Diau, *Adv. Energy Mater.*, 2018, **8**, 1701640.

202 T. H. Chowdhury, M. Akhtaruzzaman, M. E. Kayesh, R. Kaneko, T. Nodaa, J.-J. Leec and A. Islam, *Sol. Energy*, 2018, **171**, 652–657.

203 S.-P. Cho, G. Han, Y.-H. Seo, Y.-J. Noh, J.-Y. Sohn, I.-T. Hwang, J. Shin, C.-H. Jung and S.-I. Na, *Compos. Sci. Technol.*, 2021, **201**, 108548.

204 M. I. Ahmed, Z. Hussain, M. Mujahid, A. N. Khan, S. S. Javaid and A. Habib, *AIP Adv.*, 2016, **6**, 065303.

205 H. Yang, N. Liu, M. Ran, Z. He, R. Meng, M. Chen, H. Lu and Y. Yang, *J. Mater. Sci.: Mater. Electron.*, 2020, **31**, 3603–3612.

206 H. R. Mohseni, M. Dehghanipour, N. Dehghan, F. Tamaddon, M. Ahmadi, M. Sabet and A. Behjat, *Sol. Energy*, 2021, **213**, 59–66.

207 J. V. Patil, S. S. Mali, A. P. Patil, P. S. Patil and C. K. Hong, *Energy*, 2019, **189**, 116396.

208 X. Zhang, G. Ji, D. Xiong, Z. Su, B. Zhao, K. Shen, Y. Yang and X. Gao, *RSC Adv.*, 2018, **8**, 987–993.

209 R. l. Belchi, A. l. Habert, E. Foy, A. Gheno, S. Vedraine, R. m. Antony, B. Ratier, J. Bouclé and N. Herlin-Boime, *ACS Omega*, 2020, **4**, 11906–11913.

210 A. K. Kang, M. H. Zandi and N. E. Gorji, *J. Electron. Mater.*, 2020, **49**, 2289–2295.

211 A. Agresti, S. Pescetelli, B. Taheri, A. E. D. R. Castillo, L. Cin, F. Bonaccorso and A. D. Carlo, *ChemSusChem*, 2016, **9**, 2609–2619.

212 E. Nouri, Y.-L. Wang, Q. Chen, J.-J. Xu, G. Paterakis, V. Dracopoulos, Z.-X. Xu, D. Tasis, M. R. Mohammadi and P. Lianos, *Electrochim. Acta*, 2017, **233**, 36–43.

213 S. Suragtkhuu, O. Tserendavag, U. Vandandoo, A. S. R. Bati, M. Bat-Erdene, J. G. Shapter, M. Batmunkh and S. Davaasambuu, *RSC Adv.*, 2020, **10**, 9133–9139.

214 G. S. Han, J. S. Yoo, F. Yu, M. L. Duff, B. K. Kang and J.-K. Lee, *J. Mater. Chem. A*, 2017, **5**, 14733–14740.

215 T. Gatti, F. Lamberti, P. Topolovsek, M. Abdu-Aguye, R. Sorrentino, L. Perino, M. Salerno, L. Girardi, C. Marega, G. A. Rizzi, M. A. Loi, A. Petrozza and E. Menna, *Sol. RRL*, 2018, **2**, 1800013.

216 H. Li, L. Tao, F. Huang, Q. Sun, X. Zhao, J. Han, Y. Shen and M. Wang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 38967–38976.

217 A. L. Palma, L. Cinà, S. Pescetelli, A. Agresti, M. Raggio, R. Paolesse, F. Bonaccorso and A. D. Carlo, *Nano Energy*, 2016, **22**, 349–360.

218 H. Chen, Y. Hou, C. E. Halbig, S. Chen, H. Zhang, N. Li, F. Guo, X. Tang, N. Gasparini, I. Levchuk, S. Kahmann, C. O. R. Quiroz, A. Osset, S. Eigler and C. J. Brabec, *J. Mater. Chem. A*, 2016, **4**, 11604–11610.

219 J. H. Heo, D. H. Shin, D. H. Song, D. H. Kim, S. J. Lee and S. H. Im, *J. Mater. Chem. A*, 2018, **6**, 8251–8258.

220 J.-S. Yeo, R. Kang, S. Lee, Y.-J. Jeon, N. Myoung, C.-L. Lee, D.-Y. Kim, J.-M. Yun, Y.-H. Seo, S.-S. Kim and S.-I. Na, *Nano Energy*, 2015, **12**, 96–104.

221 A. Di Carlo, A. Agresti, F. Brunetti and S. Pescetelli, *J. Phys.: Energy*, 2020, **2**, 031003.

222 E. Muchuweni, E. T. Mombeshora, B. S. Martincigh and V. O. Nyamori, *Front. Mater.*, 2022, **9**, 733552.

223 E. Lamanna, F. Matteocci, E. Calabò, L. Serenelli, E. Salza, L. Martini, F. Menchini, M. Izzi, A. Agresti, S. Pescetelli, S. Bellani, A. E. D. R. o. Castillo, F. Bonaccorso, M. Tucci and A. D. Carlo, *Joule*, 2020, **4**, 865–881.

