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# Quantum interference dependence on molecular configurations for cross-conjugated systems in single-molecule junctions†

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We report a combined experimental and computational study of seven cross-conjugated enediyne derivatives functionalised with a pendant group (diphenyl, 9-fluorenyl, 9-thioxanthene or cyclohexyl) at the central alkene site, and with thiomethyl (SMe) or thioacetate, as protected thiol, (SAC) groups as anchors. Measurements of the conductance ( $G$ ) and Seebeck coefficient ( $S$ ) of gold|single-molecule|gold junctions were obtained using a modified scanning tunnelling microscope-break junction (STM-BJ) technique. It is shown that most of the molecules give multiple conductance plateaus ascribed to different molecular configurations inside the junction. The higher conductance plateaus are consistent with the aryl pendant units interacting with one of the gold electrodes, thereby circumventing transmission of electrons through the enediyne system; the lower conductance plateaus are consistent with anchoring of both of the terminal SMe or S units to the electrodes. Most of the compounds show a positive value of  $S$  in the range  $3.7\text{--}12.7\text{ }\mu\text{V K}^{-1}$  indicating electronic transport through the HOMO, while one of them presents a negative value of  $S$  ( $-6.2\text{ }\mu\text{V K}^{-1}$ ) indicating a predominance of the LUMO in the electronic transport. Theoretical calculations using density functional theory show a destructive quantum interference (DQI) feature in the gap between the highest occupied and lowest unoccupied molecular orbitals (the HOMO–LUMO gap) for the lower conductance plateaus, supporting the trends observed in the experimental data.

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## Design, System, Application

Thermoelectric materials have potential applications as large-area, flexible thermogenerators and Peltier coolers. Compared to traditional inorganic thermoelectric materials, organic counterparts are abundant, more environmentally friendly and can be processed from solution using low-temperature, high-throughput, lower-cost conditions. An efficient thermoelectric material should have a large Seebeck coefficient  $S$ , high electrical conductance  $G$  and low thermal conductivity  $\kappa$ . However, for organic molecules this combination of properties is difficult to obtain requiring optimised design of the molecular backbone, the anchor groups and the pendant groups. For example, the extent of  $\pi$ -electron conjugation in the molecular backbone can determine quantum interference effects which enhance or suppress conductance. The general utility of the present work lies in extending the range of molecules and conduction mechanisms that can be exploited to control thermoelectric properties. Measurements of the conductance and Seebeck coefficient of gold|single-molecule|gold junctions comprising enediyne molecules show that most of the molecules give multiple conductance plateaus ascribed to different molecular configurations inside the junction, which depend on the structure of the pendant groups. Theoretical calculations probe quantum interference features in the assembled system, supporting the trends observed in the experimental data. The application potential lies in the development of new organic thermoelectric materials.

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## Introduction

Experimental and theoretical studies of charge transport through single-molecules bridging two metallic electrodes (molecular junctions) are of fundamental importance in the development of molecular electronic devices such as switches, diodes, transistors, sensors and thermoelectric generators.<sup>1–5</sup> It is well established that the molecular backbone, the anchor groups and the pendant groups all play important roles in the molecular design in order to enhance electrical and thermoelectrical properties.<sup>6–8</sup> The extent of  $\pi$ -electron conjugation in the backbone is a crucial feature in



determining the electronic properties of the device.<sup>9,10</sup> The tunnelling of an electron through the orbitals localised on the molecules can be described in terms of quantum interference (QI)<sup>11</sup> which can enhance or suppress conductance.<sup>12,13</sup>

Research into organic thermoelectric materials is a rapidly expanding topic, driven by their potential applications as large-area, flexible thermogenerators and Peltier coolers. For many years, inorganic materials, such as Bi<sub>2</sub>Te<sub>3</sub> and related compounds, have set the thermoelectric performance standards.<sup>14</sup> However, they are brittle, toxic and require high temperatures to process. In contrast, organic thermoelectric materials are abundant, more environmentally friendly and can be processed from solution using low-temperature, high-throughput, lower-cost conditions.<sup>15</sup> An efficient thermoelectric material should have a large Seebeck coefficient *S*, high electrical conductance *G* and low thermal conductivity  $\kappa$ . It has been shown that destructive quantum interference (DQI) creates an anti-resonance close to the Fermi energy ( $E_F$ ),<sup>11</sup> leading to an increase in the value of *S* but compromising the enhancement of *G*. DQI is expected to appear for different molecular configurations, for example the *meta* connection of the anchoring groups or backbones<sup>16,17</sup> mechanosensitive molecules<sup>18</sup> or specific  $\pi$ -conjugation pathways, as in cross-conjugated systems.<sup>19,20</sup>

Y-shaped enediynes are classical cross-conjugated systems and their electronic and optical properties have been studied in previous reports.<sup>21–25</sup> We now present a combined experimental and theoretical study of seven cross-conjugated enediyne molecules (1–4) incorporating four different pendant groups (diphenyl, 9-fluorenyl, 9-thioxanthene or cyclohexyl) and two different anchor groups, thiomethyl (SMe) and thioacetate (SAC) as protected thiol groups that serve to assemble the molecules in gold|single-molecule|gold junctions. Note that the SAC gets deprotected once the molecule is deposited on the sample, by converting this anchor group into a thiol. The pendant groups diphenyl, 9-fluorenyl, 9-thioxanthene were chosen to provide variation in both the conjugation (and therefore energy) and rigidity of the pendant groups with the cyclohexyl group used as a non-conjugated reference. The results of scanning tunneling microscope-break junction (STM-BJ) experiments and density functional theory (DFT) calculations show the importance of these structural elements in tuning the values of the conductance and the Seebeck coefficient of the single-molecule junctions.

### Molecule synthesis

Two series of cross-conjugated enediynes with anchor groups were synthesised with diphenyl, fluorene, and thioxanthene as aromatic pendant groups (Fig. 1). The thiomethyl anchored series (1a–3a) was synthesised using a modification of Treitel's approach<sup>26</sup> involving a Sonogashira coupling between the respective diaryl-2,2-dibromoethylene and

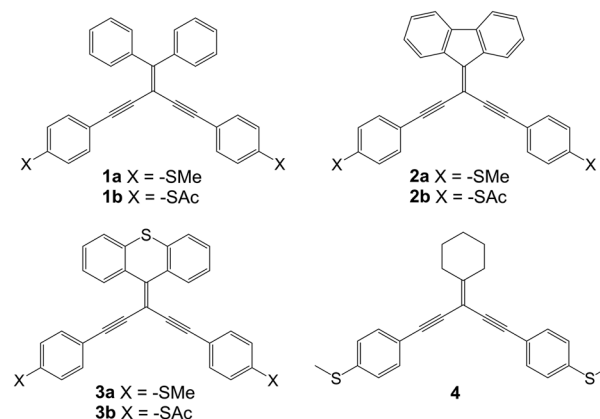


Fig. 1 Structures of compounds 1a–4 and 1b–3b used in this study.

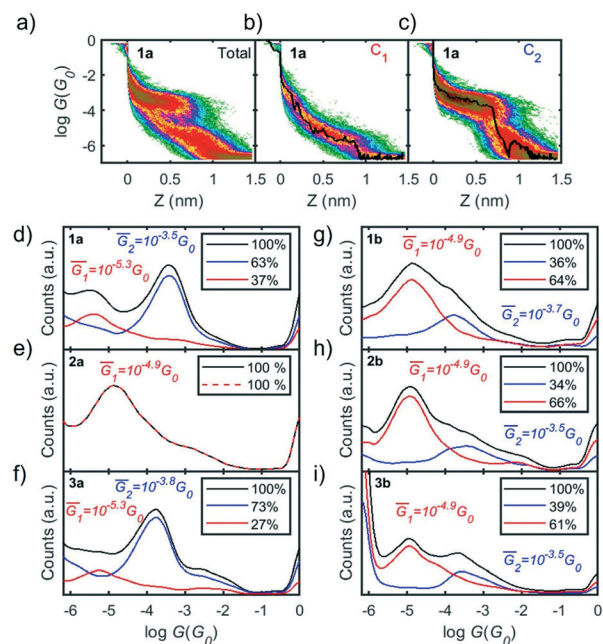
4-ethynylthioanisole. The thioacetate anchored compounds (1b–3b) were synthesised in an analogous fashion using 2-((4-ethynylphenyl)thio)ethyltrimethylsilane as the alkyne to produce the ethyltrimethylsilane protected thiols that were, in turn, deprotected by tetrabutylammonium fluoride and treated with acetyl chloride to produce the thioacetate compounds. Compound 4 with a pendant cyclohexyl unit was prepared by a Sonogashira coupling between penta-1,4-diyne-3-ylidenecyclohexane and 4-iodothioanisole. 4 was chosen to exclude the possibility of the pendant aromatic group anchoring to one of the electrodes. The detailed procedures for the synthesis of compounds 1–4 are reported in the ESI,<sup>†</sup> along with their characterization by NMR spectroscopy, mass spectrometry and elemental analysis, including the single-crystal X-ray structure of 2a.

### Quantum transport measurements

*G* and *S* measurements of molecular junctions formed with compounds 1a–3a, 1b–3b and 4 were performed using a modified home-built scanning tunnelling microscope (STM) at ambient conditions and room temperature using the STM-BJ technique.<sup>27</sup> With this technique, the sample is contacted with the STM tip and, when the tip is retracted, atomic Au contacts form before the metallic contact breaks and a molecule may be trapped between both electrodes, forming a molecular junction. By recording the current during this whole process, current–distance (*I**Z*) curves are obtained, showing a plateau in *G* values as a signature of molecular junction formation. *G* characterization was performed by collecting thousands of *I**Z* traces and eliminating those without a clear molecular plateau by using a non-supervised clustering technique.<sup>28,29</sup>

The 1D *G* histograms of the resulting *I**Z* curves for 1a–3a and 1b–3b, performed with a  $V_{\text{bias}} = 100$  mV, (shown in black in Fig. 2d–i) in general present multiple peaks, corresponding to different molecular configurations in the junction. We used the clustering technique to separate the traces corresponding to the different molecular configurations, grouping them in separate clusters. For all molecules except

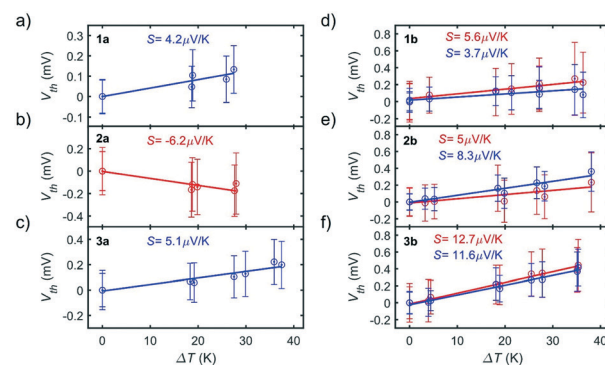




**Fig. 2** 2D histograms of  $G$  vs. distance ( $Z$ ) of molecule **1a**, (a) all the traces, (b) cluster  $C_1$  and (c) cluster  $C_2$ . A representative example of an individual trace (black line) is included in the corresponding 2D histogram for  $C_1$  and  $C_2$ . (d–i) Conductance histograms for compounds **1a–3a** (d–f), and **1b–3b** (g–i), built with all the IZ traces with a molecular plateau (black). Red and blue lines represent the corresponding histograms for clusters  $C_1$  and  $C_2$ ;  $\overline{G}_1$  and  $\overline{G}_2$  are the corresponding mean values. The percentage of traces included in each cluster is given in the legend.

for **2a**, two clusters ( $C_1$  and  $C_2$ ) were found, consisting in traces with a low- and a high-conductance plateaus. 2D histograms of  $G$  vs. distance ( $Z$ ) of the compound **1a** for all the traces and for the separated clusters  $C_1$  and  $C_2$  are shown in Fig. 2a–c, respectively, an example of individual IZ traces of  $C_1$  and  $C_2$  is included in black on top of the histograms. The corresponding 1D  $G$  histogram of each cluster is represented in Fig. 2d–i in red ( $C_1$ ) and blue ( $C_2$ ). Using a Gaussian fit we find the mean conductance value in each case, given in Fig. 2d–i as  $\overline{G}_1$  and  $\overline{G}_2$ .

Considering the series with SMe anchor groups (Fig. 2d–f), compounds **1a** and **3a** present two clusters while in contrast, compound **2a** exhibits only one cluster. To get further insight into the molecular configuration, we analysed the apparent stretching lengths ( $L_s$ )<sup>30</sup> of each cluster (see ESI† for further details and Table S3† for the values), and we found that  $C_1$  of molecule **2a** has a similar  $L_s$  as  $C_1$  of molecules **1a** and **3a**, suggesting a similar molecular configuration with a slightly higher conductance in the case of **2a**. Regarding the series with thiol anchor groups (**1b**, **2b** and **3b**) (Fig. 2g–i), similar mean conductance values both for  $\overline{G}_1$  and  $\overline{G}_2$  indicate a small effect of the pendant groups in the conductance of this series. Comparing  $L_s$  of all the compounds with two clusters (see Table S3†), we observe larger values in the case of the low-conductance cluster  $C_1$ . For this cluster the conductance values are lower and apparent molecular lengths are smaller



**Fig. 3** Seebeck coefficient ( $S$ ) of compounds **1a–3a** (a–c), and **1b–3b** (d–f). Linear regressions of all  $V_{th}$  vs.  $\Delta T$  data points included in  $C_1$  and  $C_2$  (blue and red, respectively), separated using the clustering technique. Seebeck coefficients, obtained from the slope of the regressions, are given in each panel. Empty circles and error bars represent the mean value and standard deviation from a Gaussian fit to each set of  $V_{th}$  data measured for a given  $\Delta T$ .

in the case of compounds with SMe anchor groups compared with those with thiol anchor groups, in agreement with previously reported results.<sup>31</sup>

Complementary measurements of  $G$  for compound **4** (Fig. S24 of ESI†) showed a single low-conductance peak around  $10^{-5} G_0$ . Since the cyclohexyl pendant group of **4** would not be expected to interact with the electrodes, this result suggests that the formation of the high-conductance plateaus for the other compounds (**1a–3a**, and **1b–3b**) is due to the interaction of the aromatic pendant groups with one of the electrodes.

To characterize the Seebeck coefficient, a temperature difference was established between the tip and the sample ( $\Delta T$ ) by heating up a 1 k $\Omega$  resistor placed onto the tip support and maintaining the sample at ambient temperature. While an IZ curve is performed, the thermoelectric response of the molecular junctions is measured using small current–voltage ( $IV$ ) ramps of  $\pm 10$  mV.  $G$  and the thermovoltage ( $V_{th}$ ) can be simultaneously recorded from the slope and zero crossing point of each individual  $IV$  trace, respectively (see ESI† for more information). Measurements with different  $\Delta T$  were performed to obtain a more robust statistical analysis. Note that the low-conductance values in  $C_1$  of molecules **1a** and **3a** prevented a measurement of the  $V_{th}$  of this configuration.

Applying the above-mentioned clustering technique, the measured  $V_{th}$  values were separated into clusters  $C_1$  and  $C_2$ , based on the IZ traces. The Seebeck coefficient of each cluster was then obtained from the slope of the linear regression of all  $V_{th}$  vs.  $\Delta T$  points, as shown in Fig. 3, in red and blue for  $C_1$  and  $C_2$ , respectively.

The difference between the  $S$  values of  $C_1$  and  $C_2$  for molecules with thiol anchor groups (**1b–3b**) (in Fig. 3d–f) reveals the small effect of these two different junction configurations on the thermopower. However, the different  $S$  values from molecule to molecule show the key role played by the pendant groups, with  $S$  of  $C_1$  of **3b** approximately 2.3



times larger than the value of **1b**, for example. In contrast, the measured  $S$  values for compounds **1a** and **3a** (which have SMe anchor groups) are very similar (Fig. 3a and c), reflecting a smaller effect of the pendant groups.

The positive sign of  $S$  for all molecules, except **2a**, indicates transport mainly through the HOMO.<sup>7</sup> The negative sign of  $S$  for **2a** is a signature of LUMO predominance in the electronic transport, as previously reported for SMe anchor groups.<sup>16,32</sup>

To get further insight into the molecular configurations in the junction and the origin of the different signs of  $S$  for the different clusters we performed quantum transport calculations.

### Theoretical calculations

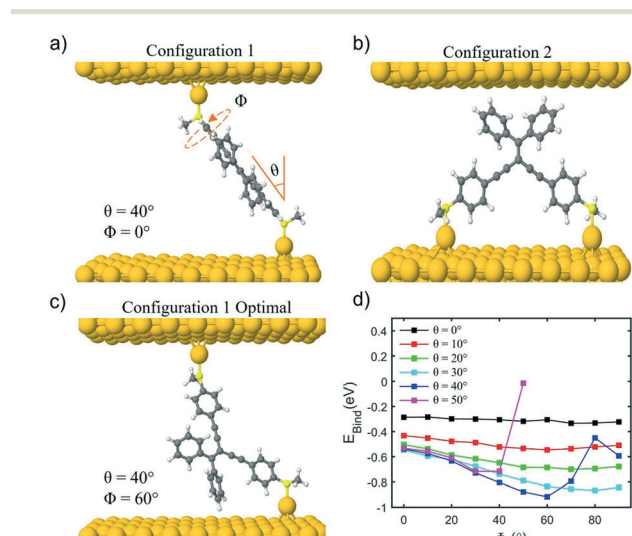
The density functional code SIESTA<sup>33</sup> and transport code GOLLUM<sup>34</sup> were used to calculate the conductance and Seebeck coefficients of the molecules. First, the optimum geometry of each molecule was found and then the molecule was attached to gold electrodes. Assuming both types of anchor groups (SMe and thiol) attach to a surface adatom, two different binding configurations were investigated, namely, with one anchor group connected to each electrode (configuration 1) or with both anchor groups connected to the same electrode (configuration 2) (see Fig. 4a and b). This second configuration was considered based on the 'bent' backbone of the compounds.

Considering configuration 1, the optimal binding geometry with respect to the binding energy was calculated as a function of two different parameters, in addition to the

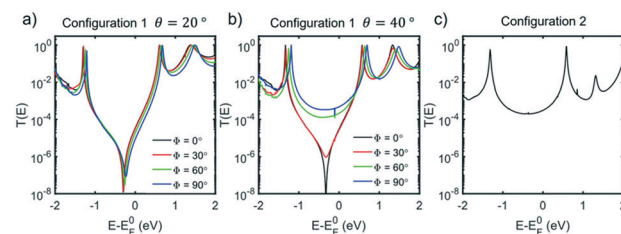
Au-S distance (which was found to be 2.3 Å for S and 2.4 Å for SMe). These parameters are: (i) the tilt angle  $\theta$  and (ii) the rotation about the axis of the molecule  $\Phi$  (Fig. 4a). For example, in the case of a tilt angle of 90°, at  $\Phi = 0^\circ$  the molecule would lie parallel to the gold surface. The binding energy  $E_{\text{Bind}}$  of molecule **1a** was then evaluated for these parameters in Fig. 4d which shows that as the molecule is tilted ( $\theta > 20^\circ$ ) the binding energy increases in magnitude as the value of  $\Phi$  is increased; this is due to the increased interaction between the pendant group and the gold surface. In the case of **1a** the optimum binding geometry is  $\theta = 40^\circ$  and  $\Phi = 60^\circ$  with a binding energy of -0.92 eV. The binding energy of molecules **2-4** and **1-3b** shows similar trends (Fig. S34-S40†).

Cross-conjugation has been shown to lead to quantum interference in molecular junctions<sup>35</sup> and typically leads to destructive quantum interference (DQI) features. One simple method to determine the nature of the interference is the orbital product rule<sup>36</sup> which predicts that destructive interference will occur when the orbital products of the contact site have the same sign for the HOMO and LUMO, and constructive interference will occur when they have the opposite sign. The HOMO and LUMO of all compounds are shown in Fig. S27-S33 of ESI† (the sign of the wave function is denoted blue for negative, red for positive) and the wavefunction on both the contact atoms has the same sign, for both HOMO and LUMO, so the product is always positive, indicating destructive interference.

The electron transport through molecule **1a** was then calculated to see if a DQI feature occurs. Fig. 5a and b shows the transmission coefficient  $T(E)$  for two different tilt angles ( $\theta = 20^\circ$  and  $40^\circ$ ). In the case of  $\theta = 20^\circ$  the transmission shows a DQI feature in the HOMO-LUMO gap as predicted by the orbital product rule. Changing the rotation angle  $\Phi$  at this tilt angle has only a small effect on the transmission. At a tilt angle of  $\theta = 40^\circ$  the transmission shows a DQI feature for  $\Phi = 0^\circ$ , at this rotation angle the pendant group is parallel to the gold surface (Fig. 4a) and so there is only a weak interaction between them. As the molecule is rotated the DQI feature in the transmission is suppressed ( $\Phi > 60^\circ$ ); this is due to the increased interaction between the pendant group and the gold surface which causes the orbital product rule to break down as there are now multiple contact sites between



**Fig. 4** a) Molecular junction configuration 1, with one anchor group connected to each electrode, with tilt angle ( $\theta = 40^\circ$ ) and rotation angle ( $\Phi = 0^\circ$ ) for compound **1a**. b) Molecular junction configuration 2, with both anchor groups connected to the same electrode, for compound **1a**. c) Optimum binding geometry of compound **1a** for configuration 1 with  $\theta = 40^\circ$  and  $\Phi = 60^\circ$ . d) Binding energy ( $E_{\text{Bind}}$ ) of **1a** on gold(111) surface as a function of rotation angle  $\Phi$  for tilt angles  $\theta$  between 0 and  $50^\circ$ .



**Fig. 5** Zero-bias transmission coefficient  $T(E)$  of **1a** for values of  $\Phi$  between 0 and  $90^\circ$ , for configuration 1, at tilt angle  $\theta = 20^\circ$  (a) and  $\theta = 40^\circ$  (b). (c) Zero-bias transmission coefficient  $T(E)$  of **1a** for configuration 2.





the gold surface and the molecule. Fig. S41–S47† show the same general trend for molecules **1a**, **2a**, **4** and **1b–3b**.

The fact that the binding energy increases when the pendant group interacts with the gold suggests a contact geometry where the top electrode is attached to this substituent as shown in Fig. 4b (the so-called configuration 2). The optimum binding distance between the flat surface of the lead and the pendant group of **1a** was calculated, then the transmission was calculated as shown in Fig. 5c (see Fig. S48–S52† for the corresponding transmissions for configuration 2 of compounds **2a–3a** and **1b–3b**; the cyclohexyl group of **4** is assumed not to bind to the gold surface). The transmission was found to be larger for configuration 2 than for configuration 1 in all cases and, together with the optimum binding energies, this allows us to associate molecular configuration 1 to cluster  $C_1$  and configuration 2 to cluster  $C_2$  (based on the experimental conductance and  $L_s$  values for each compound). The theoretical distance between the electrodes for configuration 1 and configuration 2 (Table S6†) follows the trends of the clusters  $C_1$  and  $C_2$  (Table S4†) with the separation of configuration 1 being larger than that of configuration 2.

The values of  $G$  and  $S$  of all compounds for configuration 1 were then evaluated (see Fig. S48–S54†) for several values of  $\Phi$  and  $\theta$  and at two values of the Fermi energy, namely,  $E_F^0$  and  $E_F^1$ , where  $E_F^0$  is the DFT predicted Fermi energy and  $E_F^1$  is chosen to give the best agreement with the magnitude of the measured values. This correction is applied because the alignment of the frontier molecular energy levels in the junction relative to  $E_F^0$  can show significant errors in DFT.<sup>37,38</sup> For the SMe anchor groups (compounds **1a–4**) the Fermi energy was chosen to be  $E_F^1 = E_F^0 - 0.5$  eV and for the S anchor (compounds **1b–3b**),  $E_F^1 = E_F^0 + 0.4$  eV. Table 1 shows the comparison between experiment and theory for all the molecules studied, considering these values of  $E_F^1$  (also applied to the transmissions of configuration 2). In general,  $G$  and  $S$  values show similar trends, however the differences in magnitude could be due to experimental conditions that have not been included in the theoretical model such as the junction environment. In the case of **2a** the negative value of  $S$  can be attributed to the lower lying LUMO level for this molecule; the equivalent is not found for molecule **2b** as the S anchors pin the HOMO resonance to the Fermi energy.<sup>39</sup> Overall, the relatively small values of  $S$  observed in these

molecules can be explained by the interaction of the pendant group with one of the gold electrodes suppressing the DQI feature which causes the high value of  $S$  at low tilt angles, as previously discussed.

Finally, to explore the possibility of having fully stretched molecules in the junction, complementary conductance measurements with a bias voltage of 600 mV were also performed, allowing us to reach lower conductance ranges. Using again the clustering technique, for compounds **1a–3a** we were able to identify only the clusters observed with 100 mV bias voltage (*i.e.*,  $C_1$  and  $C_2$ ) while for molecules **1b–3b** a third cluster,  $C_3$ , was also observed (see Fig. S25 of the ESI†). This new cluster presents a lower conductance mean value ( $\bar{G}_3$ ) and longer  $L_s$  than  $C_1$  and  $C_2$  (see Table S4 of the ESI†). This suggests that the molecular configuration in these cases might be, indeed, a completely stretched one, without interaction between the pendant groups and the electrodes. This is also in agreement with the theoretical  $G$  values for small tilt angles, which in the case of compounds **1b–3b** (Fig. S55–S59†) is at least an order of magnitude lower than the value for configuration 1.

## Conclusions

In summary, we have synthesized seven cross-conjugated enediyne derivatives that are functionalised with a pendant group (diphenyl, 9-fluorenyl, 9-thioxanthene or cyclohexyl) at the central alkene site, and with thiomethyl (SMe) or SAc (as protected thiol) groups attached as anchors. Their single-molecule conductance and Seebeck coefficient have been measured by STM-BJ techniques and corresponding theoretical calculations have been performed using DFT. Notably, two different conductance plateaus are observed for all the studied compounds, except for **2a**. The high-conductance plateaus are consistent with anchoring of both of the terminal SMe or thiol units to a single gold electrode and the aryl pendant units interacting with the other electrode, thereby avoiding transmission of electrons through the enediyne system. On the other hand, the low-conductance plateaus are consistent with anchoring of one terminal SMe or thiol units to each electrode – a configuration that may lead to a destructive quantum interference (DQI) feature in the HOMO–LUMO gap, depending on the rotation angle of the molecule. The small difference between  $S$  of the different conductance plateaus suggests the suppression of the DQI in these molecular junctions as a consequence of the interaction of the pendant groups with the electrodes. Compound **2a** with SMe anchors groups presents a different behavior in all the measurements, showing only a single plateau, with a conductance value and apparent stretching length consistent with the configuration with one terminal unit anchored to each electrode. Additionally, **2a** shows a negative  $S$  value, indicating a LUMO predominance in the electronic transport; all the other compounds give a positive  $S$  value indicating HOMO-dominated transport. Overall, this study sheds new light on

**Table 1** Comparison of the experimental and theoretical values for the conductance  $G$  and Seebeck coefficient  $S$

Compound	$\log G_1(G_0)$		$\log G_2(G_0)$		$S_1 (\mu V K^{-1})$		$S_2 (\mu V K^{-1})$	
	Exp	Theo	Exp	Theo	Exp	Theo	Exp	Theo
<b>1a</b>	−5.3	−4.2	−3.5	−3.9		5.1	4.2	3.0
<b>2a</b>	−4.9	−4.5		−3.7	−6.2	−10.2		−6.5
<b>3a</b>	−5.3	−4.2	−3.8	−3.5		4.1	5.1	4.3
<b>4</b>	−5.0	−5.2				5.7		
<b>1b</b>	−4.9	−4.4	−3.7	−3.2	3.7	3.8	5.6	35.5
<b>2b</b>	−4.9	−4.0	−3.5	−3.2	8.3	11.5	5.0	23.0
<b>3b</b>	−4.9	−3.7	−3.5	−3.0	11.6	5.8	12.7	31.4



the range of molecules and mechanisms that can be exploited to control the Seebeck coefficient in single molecules, representing a step towards producing highly efficient thin-film thermoelectric devices for practical applications.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- R. J. Nichols and S. J. Higgins, *Annu. Rev. Anal. Chem.*, 2015, **8**, 389–417.
- P. T. Mathew and F. Fang, *Engineering*, 2018, **4**, 760–771.
- D. Xiang, X. Wang, C. Jia, T. Lee and X. Guo, *Chem. Rev.*, 2016, **116**, 4318–4440.
- T. A. Su, M. Neupane, M. L. Steigerwald, L. Venkataraman and C. Nuckolls, *Nat. Rev. Mater.*, 2016, **1**, 16002.
- J. Cuevas and E. Scheer, *Molecular Electronics: An Introduction to Theory and Experiment*, World Scientific, 2nd edn, 2017.
- W. B. Chang, C. K. Mai, M. Kotiuga, J. B. Neaton, G. C. Bazan and R. A. Segalman, *Chem. Mater.*, 2014, **26**, 7229–7235.
- L. Rincón-García, C. Evangeli, G. Rubio-Bollinger and N. Agrait, *Chem. Soc. Rev.*, 2016, **45**, 4285–4306.
- L. Rincón-García, A. K. Ismael, C. Evangeli, I. Grace, G. Rubio-Bollinger, K. Porfyrakis, N. Agrait and C. J. Lambert, *Nat. Mater.*, 2016, **15**, 289–293.
- L. J. O'Driscoll and M. R. Bryce, *Nanoscale*, 2021, **13**, 10668–10711.
- H. Valkenier, C. M. Guédon, T. Markussen, K. S. Thygesen, S. J. van der Molen and J. C. Hummelen, *Phys. Chem. Chem. Phys.*, 2014, **16**, 653–662.
- C. J. Lambert, *Chem. Soc. Rev.*, 2015, **44**, 875–888.
- D. M. Cardamone, C. A. Stafford and S. Mazumdar, *Nano Lett.*, 2006, **6**, 2422–2426.
- X. Li, Z. Tan, X. Huang, J. Bai, J. Liu and W. Hong, *J. Mater. Chem. C*, 2019, **7**, 12790–12808.
- J. Wei, L. Yang, Z. Ma, P. Song, M. Zhang, J. Ma, F. Yang and X. Wang, *J. Mater. Sci.*, 2020, **55**, 12642–12704.
- M. Lindorf, K. A. Mazzio, J. Pflaum, K. Nielsch, W. Brütting and M. Albrecht, *J. Mater. Chem. A*, 2020, **8**, 7495–7507.
- G. Yzambart, L. Rincón-García, A. A. Al-Jobory, A. K. Ismael, G. Rubio-Bollinger, C. J. Lambert, N. Agrait and M. R. Bryce, *J. Phys. Chem. C*, 2018, **122**, 27198–27204.
- I. M. Grace, G. Olsen, J. Hurtado-Gallego, L. Rincón-García, G. Rubio-Bollinger, M. R. Bryce, N. Agrait and C. J. Lambert, *Nanoscale*, 2020, **12**, 14682–14688.
- D. Stefani, K. J. Weiland, M. Skripnik, C. Hsu, M. L. Perrin, M. Mayor, F. Pauly and H. S. J. van der Zant, *Nano Lett.*, 2018, **18**, 5981–5988.
- A. Alanazy, E. Leary, T. Kobatake, S. Sangtarash, M. T. González, H. W. Jiang, G. R. Bollinger, N. Agrait, H. Sadeghi, I. Grace, S. J. Higgins, H. L. Anderson, R. J. Nichols and C. J. Lambert, *Nanoscale*, 2019, **11**, 13720–13724.
- S. Medina Rivero, P. García Arroyo, L. Li, S. Gunasekaran, T. Stuyver, M. J. Mancheño, M. Alonso, L. Venkataraman, J. L. Segura and J. Casado, *Chem. Commun.*, 2021, **57**, 591–594.
- M. A. Christensen, E. A. della Pia, J. Houmøller, S. Thomsen, M. Wanko, A. D. Bond, A. Rubio, S. Brøndsted Nielsen and M. Brøndsted Nielsen, *Eur. J. Org. Chem.*, 2014, **2014**, 2044–2052.
- L. O. Jones, M. A. Mosquera, B. Fu, G. C. Schatz, T. J. Marks and M. A. Ratner, *Nano Lett.*, 2019, **19**, 8956–8963.
- D. Q. Andrews, G. C. Solomon, R. P. van Duyne and M. A. Ratner, *J. Am. Chem. Soc.*, 2008, **130**, 17309–17319.
- A. Singh, A. K. Pati and A. K. Mishra, *Phys. Chem. Chem. Phys.*, 2018, **20**, 14889–14898.
- A. A. Kocherzhenko, L. D. A. Siebbeles and F. C. Grozema, *J. Phys. Chem. Lett.*, 2011, **2**, 1753–1756.
- N. Treitel, L. Eshdat, T. Sheradsky, P. M. Donovan, R. R. Tykwinski, L. T. Scott, H. Hopf and M. Rabinovitz, *J. Am. Chem. Soc.*, 2006, **128**, 4703–4709.
- C. Evangeli, K. Gillemot, E. Leary, M. T. González, G. Rubio-Bollinger, C. J. Lambert and N. Agrait, *Nano Lett.*, 2013, **13**, 2141–2145.
- L. A. Zotti, B. Bednarz, J. Hurtado-Gallego, D. Cabosart, G. Rubio-Bollinger, N. Agrait and H. S. J. van der Zant, *Biomolecules*, 2019, **9**, 580–592.
- D. Cabosart, M. el Abbassi, D. Stefani, R. Frisenda, M. Calame, H. S. J. van der Zant and M. L. Perrin, *Appl. Phys. Lett.*, 2019, **114**, 143102.
- C. R. Arroyo, E. Leary, A. Castellanos-Gómez, G. Rubio-Bollinger, M. T. González and N. Agrait, *J. Am. Chem. Soc.*, 2011, **133**, 14313–14319.
- R. Frisenda, S. Tarkuç, E. Galán, M. L. Perrin, R. Eelkema, F. C. Grozema and H. S. J. van der Zant, *Beilstein J. Nanotechnol.*, 2015, **6**, 1558–1567.
- A. Ismael, X. Wang, T. L. R. Bennett, L. A. Wilkinson, B. J. Robinson, N. J. Long, L. F. Cohen and C. J. Lambert, *Chem. Sci.*, 2020, **11**, 6836–6841.
- J. M. Soler, E. Artacho, J. D. Gale, A. García, J. Junquera, P. Ordejón and D. Sánchez-Portal, *J. Phys.: Condens. Matter*, 2002, **14**, 2745–2779.
- J. Ferrer, C. J. Lambert, V. M. García-Suárez, D. Z. Manrique, D. Visontai, L. Oroszlany, R. Rodríguez-Ferradás, I. Grace, S. W. D. Bailey, K. Gillemot, H. Sadeghi and L. A. Algharagholy, *New J. Phys.*, 2014, **16**, 093029.



- 35 L. J. O'Driscoll and M. R. Bryce, *Nanoscale*, 2021, **13**, 1103–1123.
- 36 C. J. Lambert and S. X. Liu, *Chem. – Eur. J.*, 2018, **24**, 4193–4201.
- 37 J. B. Neaton, M. S. Hybertsen and S. G. Louie, *Phys. Rev. Lett.*, 2006, **97**, 216405.
- 38 S. Y. Quek, L. Venkataraman, H. J. Choi, S. G. Louie, M. S. Hybertsen and J. B. Neaton, *Nano Lett.*, 2007, **7**, 3477–3482.
- 39 R. Stadler and K. W. Jacobsen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2006, **74**, 161405.

