

Cite this: *Mater. Adv.*, 2022, **3**, 8250

Vapor-phase hydrothermal construction of defective MoS₂ for highly selective electrocatalytic hydrogenation of cinnamaldehyde†

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Electrocatalytic selective hydrogenation of α,β -unsaturated aldehydes with water as the hydrogen donor is of great significance to produce fine chemicals. Here, we propose a facile vapor-phase hydrothermal (VPH) method to directly grow defective MoS₂ nanoparticles on commercial carbon fiber cloth (MoS₂/CFC) for electrocatalytic selective hydrogenation of cinnamaldehyde (CAL). The as-prepared MoS₂/CFC exhibited the highest CAL conversion of 88.8% with a maximum TOF value of 12.8 mmol mmol_{MoS₂}⁻¹ h⁻¹ at -0.7 V vs. RHE. The high selectivity of CAL hydrogenation could be realized by adjusting the applied potentials, C=O hydrogenation product cinnamyl alcohol (COL) was easier to form under lower applied potentials, whereas C=C hydrogenation product hydrocinnamaldehyde (HCAL) was easier to achieve under higher applied potentials. Density functional theory (DFT) calculation findings demonstrated that the Mo vacancies in the defective MoS₂/CFC facilitated the adsorption of hydrogen atoms, which benefited the generation of H_{ads} and the proceeding of electrocatalytic hydrogenation. Additionally, the existence of Mo vacancies made the C=O group more active than the C=C group, and reduced the reaction energy of CAL hydrogenation toward COL. Combined with the experimental results and calculation data, we speculate that the real catalytic active center of the defective MoS₂/CFC is the defective configuration of Mo vacancies. Furthermore, the high conversion and selectivity of furfural and benzaldehyde hydrogenation confirmed the universality of the defective MoS₂/CFC catalyst in electrocatalytic hydrogenation reduction of unsaturated and saturated aldehydes.

Received 22nd July 2022,
Accepted 27th August 2022

DOI: 10.1039/d2ma00842d

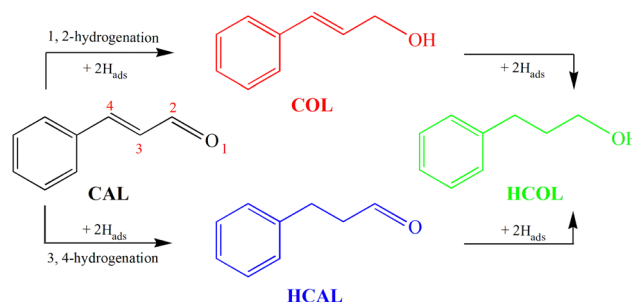
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1. Introduction

As an old and rich discipline, electrocatalysis has been shown to be a powerful method to perform organic synthesis reactions under green, mild and safe conditions.¹ Among which, electrocatalytic hydrogenation (ECH) is an economical and environmentally friendly approach to realize the hydrogenation of organics containing unsaturated chemical bonds using water as the hydrogen donor.^{2–4} ECH utilizes chemisorbed hydrogen (H_{ads}) via proton or water reduction,^{5,6} which avoids the difficult activation of low-soluble H₂ in solvents, and thereby the elevated pressure and temperature that are usually desired in thermocatalysis.^{7,8}

The selective hydrogenation of α,β -unsaturated aldehydes into the corresponding unsaturated alcohols is significant for

the production of fine chemicals.⁹ As a classical model of α,β -unsaturated aldehydes, the hydrogenation of cinnamaldehyde (CAL) has been widely investigated.^{10–14} As shown in Scheme 1, the CAL hydrogenation products include not only the desired half-hydrogenated cinnamyl alcohol (COL), but also half-hydrogenated hydrocinnamaldehyde (HCAL) and full-hydrogenated hydrocinnamyl alcohol (HCOL). However, the two adjacent unsaturated C=C and C=O bonds of CAL make the



Scheme 1 The hydrogenation pathways of cinnamaldehyde (CAL).

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† Electronic supplementary information (ESI) available: Additional SEM images and electrochemical. See DOI: <https://doi.org/10.1039/d2ma00842d>

hydrogenation reaction complex and uncontrollable.¹⁵ From the perspective of thermodynamics, chemical reactions tend to occur on the C=C bond because the C=C bond energy (615 kJ mol⁻¹) is lower than the C=O bond energy (715 kJ mol⁻¹), and the change of free energy enthalpy of C=C bond hydrogenation is 35 kJ mol⁻¹ smaller than that of C=O bond hydrogenation. Therefore, achieving highly selective hydrogenation of the C=O bond is extremely challenging.^{16,17} Recently, Zhang and co-workers developed a facile strategy to synthesize hollow CoS₂ and CoS_{2-x} nanocapsules for electrocatalytic selective transfer hydrogenation of CAL with water to synthesize HCAL and HCOL.⁹ Unfortunately, this report did not realize the high selectivity of C=O hydrogenation. Wei and co-workers reported the highly selective electrochemical hydrogenation of CAL to COL on the RuO₂-SnO₂-TiO₂/Ti electrode.¹⁸ However, a fly in the ointment is the introduction of the precious metal Ru, which limits the large-scale production and application of electrocatalysts.

ECH is challenging due to the existence of the competing reaction hydrogen evolution reaction (HER),² which consumes H_{ads} and reduces the faradaic efficiency of ECH.¹⁹ Hence, electrocatalysts that afford active and selective ECH but prohibit the HER are highly desirable.^{20,21} Enhancing the adsorption of carbonyl is a widely used strategy to improve C=O hydrogenation.¹⁸ Precious metals (*e.g.*, Ag, Pd, Pt, *etc.*) display the active ECH of carbonyl with a low onset potential,^{19,22} however, they suffer from poor faradaic efficiency, high cost, and low earth-abundance.²³ Thus, the development of efficient, high-performance, and non-precious electrocatalysts is urgently required.

Molybdenum disulfide (MoS₂), a conventional HER electrocatalyst,²⁴⁻²⁶ has strengthened surface chemisorption that is expected to stabilize H_{ads} toward hydrogenation rather than self-recombination to H₂. Furthermore, MoS₂ can effectively activate unsaturated bonds (*e.g.*, C=O), enabling a rapid elementary step on the catalyst surface.²³ For MoS₂, the preferentially exposed basal plane is the thermodynamically stable (002) plane rather than the active edge planes.^{27,28} To circumvent this disadvantage, the engineering defect structure on the basal planes can be expected to increase the exposure of active edge sites by forming cracks on the surfaces,²⁹ which may dramatically improve the electrocatalytic performance.

Herein, we report a vapor-phase hydrothermal (VPH) approach to *in situ* grow defective MoS₂ nanoparticles using commercial carbon fiber cloth (CFC) with controllably adsorbed Mo⁶⁺ as the substrate and excess thiourea as the sulfur source at 180 °C for 24 h (MoS₂/CFC). The as-prepared MoS₂/CFC without further treatment was directly used as the electrode for ECH in 0.1 M phosphate buffer electrolyte (pH = 7.0), and exhibited highly selective hydrogenation of CAL to HCAL and COL. The high selectivity of CAL hydrogenation could be realized by adjusting the applied potentials, the hydrogenation of CAL to COL was easier to achieve under lower applied potential, whereas the hydrogenation of CAL to HCAL was easier to achieve under higher applied potential. DFT calculations demonstrated that the Mo vacancies in the defective MoS₂ facilitated the adsorption of hydrogen atoms, which benefited the generation

of H_{ads} and the proceeding electrocatalytic hydrogenation. Besides, the Mo vacancies made the C=O group more active than the C=C group, and reduced the reaction energy of CAL hydrogenation toward COL. Furthermore, the high conversion and selectivity of other unsaturated aldehyde such as FAL and saturated aldehyde such as benzaldehyde hydrogenation confirmed the universality of the defective MoS₂/CFC catalyst in the electrocatalytic hydrogenation reduction of aldehydes.

2. Experimental section

2.1 Materials

Sodium molybdate (Na₂MoO₄·2H₂O), thiourea, sodium hydrogen phosphate (Na₂HPO₄·12H₂O), sodium dihydrogen phosphate (NaH₂PO₄·2H₂O), tetrahydrofuran and anhydrous ethanol were purchased from Sinopharm Chemical Reagent Co., Ltd (Shanghai, China). Cinnamaldehyde (CAL), furfural (FAL) and benzaldehyde were purchased from Aladdin Ltd (Shanghai, China). All the chemicals were used as received without further purification. Commercial carbon fiber cloth (CFC) was purchased from Shanghai Hesen Electric Co., Ltd, which was washed with deionized water and anhydrous ethanol several times, respectively, then dried in air and cut into different sizes for further use.

2.2 Vapor-phase hydrothermal construction of defective MoS₂ on CFC

In a typical synthesis, 0.50 g Na₂MoO₄·2H₂O powders were fully dissolved in 50 ml deionized water, and then a piece of cleaned CFC substrate (3.0 cm × 3.0 cm) was immersed in the above solution for 24 h to adequately adsorb Mo⁶⁺. Subsequently, the Mo⁶⁺-adsorbed CFC was dried at room temperature and then placed on a Teflon holder. The VPH reaction was performed in a Teflon-lined stainless steel autoclave (100 ml) with Mo⁶⁺-adsorbed CFC located above 20 ml thiourea solution (0.35 g thiourea/20 ml deionized water).³⁰⁻³² The VPH reaction was performed at 180 °C for 24 h. After the autoclave was cooled down to room temperature, the product was collected and immersed into 0.1 M HCl for 2 h to remove surface adsorbed ammonia generated from the decomposition of thiourea, then rinsed with deionized water several times, and dried at 60 °C under vacuum. The obtained product was denoted as MoS₂/CFC. The loading amount of MoS₂ on the CFC substrate was determined by inductively coupled plasma atomic emission spectroscopy (ICP-AES, 6300, Thermo Fisher Scientific, USA) after digestion by aqua regia through hydrothermal reaction under 180 °C for 48 h, and revealed a MoS₂ loading of 0.56 mg cm⁻².

For comparison, bulk MoS₂ was synthesized through a liquid-phase hydrothermal reaction. 0.50 g Na₂MoO₄·2H₂O and 0.35 g thiourea were fully dissolved in 20 ml deionized water. The hydrothermal reaction was performed at 180 °C for 24 h.

2.3 Characterizations

The morphological properties of the samples were investigated using a field emission scanning electron microscope (FESEM,



Hitachi SU8020) operated at an accelerating voltage of 3.0 kV. The microstructures of the samples were examined by a high-resolution transmission electron microscope (HRTEM, FEI Tecnai G2 F30 S-TWIN) with an acceleration voltage of 300 kV. X-Ray diffraction (XRD) patterns of the samples were recorded on a Philips X-Pert Pro X-ray diffractometer with Cu K α radiation ($\lambda_{\text{K}\alpha 1} = 1.5418 \text{ \AA}$). XPS analysis of the samples was performed on an ESCALAB 250 X-ray photoelectron spectrometer (Thermo, America) equipped with Al K $\alpha_{1,2}$ monochromatized radiation at 1486.6 eV X-ray source. ICP measurements were performed by inductively coupled plasma atomic emission spectroscopy (ICP-AES, 6300, Thermo Fisher Scientific, USA).

2.4 Electrochemical measurements

The electrochemical measurements were performed with a CHI 760E electrochemical workstation (CH Instruments, Inc., Shanghai, China), and a typical three-electrode divided H-type cell was used. The cathodic compartment and the anodic compartment were divided by a piece of Nafion 115 proton exchange membrane. Both the compartments contained 20 ml 0.1 M phosphate buffer electrolyte (pH = 7.0). If without extra statement, the electrolyte in the cathodic compartment maintained continuous stirring during all the electrochemical measurements. An Ag/AgCl electrode as the reference electrode was placed in the cathodic compartment, and a piece of pre-cleaned CFC (2.0 cm \times 3.0 cm) as the counter electrode was placed in the anodic compartment. The as-prepared MoS₂/CFC (1.0 cm \times 2.0 cm) was directly used as the working electrode, and the effective geometric surface area of the working electrode was maintained at 1.0 cm \times 1.0 cm. For bulk MoS₂ powders, the working electrodes were prepared as follows: 6.0 mg as-prepared MoS₂ powders were dispersed in 1000 μl of the mixed solution comprising 500 μl of deionized water, 490 μl of ethanol, and 10 μl of 5 wt% Nafion solution. The suspension was sonicated for 20 min to obtain a homogeneous ink. Then, 100 μl of MoS₂ ink was dropped onto a cleaned CFC (1.0 \times 2.0 cm, the MoS₂ covering geometric area was 1.0 \times 1.0 cm²) and then dried at room temperature. The areal loading mass of MoS₂ on the CFC was about 0.60 mg.

Electrochemical impedance spectroscopy (EIS) measurements were carried out in a frequency range of 100 kHz to 0.01 Hz at an open circuit voltage with an amplitude of 5 mV. Linear sweep voltammetry (LSV) curves were obtained at a scan rate of 5.0 mV s⁻¹.

0.5 ml tetrahydrofuran containing 33 μl (0.25 mmol) cinnamaldehyde was added into the cathodic compartment and stirred for 10 min until the electrolyte was emulsified. LSV and EIS measurements followed. All the potentials were converted to the reversible hydrogen electrode (RHE) scale through calibration with the following Nernst equation: $E (V \text{ vs. RHE}) = E (V \text{ vs. Ag/AgCl}) + 0.059 \times \text{pH} + 0.197$. The hydrogenation process was evaluated by chronoamperometry under different constant potentials for 5 h.

Another two aldehyde molecules, furfural (FAL) and benzaldehyde, were also investigated. For furfural, 21 μl (0.25 mmol) furfural was directly added into the cathodic compartment.

For benzaldehyde, 0.5 ml tetrahydrofuran containing 26 μl (0.25 mmol) benzaldehyde was added into the cathodic compartment. Tetrahydrofuran was used because of its excellent solubility toward organic compounds and good stability during electrolysis.

2.5 Product analysis

After electrolysis reactions, 5.0 ml electrolyte solution was taken from the cathodic compartment and 3.0 ml ethyl acetate was added to extract the organics. The extracted organics were subsequently analyzed by gas chromatography-mass spectrometry (GC-MS, Agilent GC7890 + MS5977B). For cinnamaldehyde and benzaldehyde, the initial GC temperature was 100 $^{\circ}\text{C}$ for 2 min, and then increased at 10 $^{\circ}\text{C min}^{-1}$ to 300 $^{\circ}\text{C}$ where it was held for 2 min. For furfural, the initial GC temperature was 30 $^{\circ}\text{C}$ for 2 min, and then increased at 5 $^{\circ}\text{C min}^{-1}$ to 300 $^{\circ}\text{C}$ where it was held for 2 min.

The conversion (%) of the reactant was calculated based on the following equation:

$$\text{Conversion}(\%) = \frac{\text{Mole of reactant converted}}{\text{Mole of initial reactant}} \times 100\% \quad (1)$$

The product selectivity (%) was calculated by the following equation:

$$\text{Selectivity}(\%) = \frac{\text{Mole of target product formed}}{\text{Mole of all products formed}} \times 100\% \quad (2)$$

The faradaic efficiency (FE, %) of products formation was calculated using the following equation:

$$\text{FE}(\%) = \frac{\text{Mole of products formed}}{\text{Total charge passed}/(n \times F)} \times 100\% \quad (3)$$

where n is the number of electrons transferred for products formation (here $n = 2$), and F is the Faraday constant (96 485 C mol⁻¹).

The turnover frequency (TOF) values of MoS₂/CFC catalysts in electrocatalytic hydrogenation were calculated with the following equation while assuming all MoS₂ components are available active sites and involved in the electrocatalytic hydrogenation process:

$$\begin{aligned} \text{TOF}(\text{mmol mmol}_{\text{MoS}_2}^{-1} \text{ h}^{-1}) \\ = \frac{\text{Mole of reactant converted (mmol)}}{\frac{m_{\text{MoS}_2}(\text{mg})}{M_{\text{MoS}_2}(\text{mg mmol}^{-1})} \times t(\text{h})} \end{aligned} \quad (4)$$

where m_{MoS_2} is the MoS₂ loading amount of 0.56 mg, M_{MoS_2} is the relative molecular mass of MoS₂ (160 mg mmol⁻¹), and t is the electrolysis reaction time (5 h).

2.6 Theoretical calculations

All the first-principle calculations were performed within the framework of DFT as implemented in the Vienna *Ab initio* Simulation Package (VASP).^{33–35} The projector augmented wave (PAW) method has been used to describe the inert core electrons.³⁶ A cut off energy of 400 eV was used for the



expansion of the wave functions. The electronic exchange–correlation effects were described with Perdew–Burke–Ernzerhof generalized gradient approximation (PBE–GGA) functional.³⁷ The basic carbon model was built using a periodic graphene structure containing 72 C atoms. A slab of MoS₂(002) surface without defects, with Mo vacancies or Mo/S vacancies was built for the adsorption calculations, respectively. A vacuum of 20 Å in the z-direction is used. The gamma (Γ) centered $2 \times 2 \times 1$ Monkhorst-Pack k -point sampling was used throughout.³⁸ The Fermi level was slightly broadened using a Fermi–Dirac smearing of 50 meV. All relaxations were carried out until the force of the system converges to 0.05 eV Å⁻¹. The van der Waals interactions were considered using the DFT–D3 empirical correction.

3. Results and discussion

In this work, commercial carbon fiber cloth (Fig. S1, ESI[†]) with controllably adsorbed Mo⁶⁺ was used as the substrate for VPH growth of MoS₂ nanoparticles (MoS₂/CFC). After the VPH reaction at 180 °C for 24 h (Fig. S2, ESI[†]), the obtained products still exhibited the configuration from the original CFC framework (Fig. S3, ESI[†]). Fig. 1a shows a single fiber in which the growth occurs on the surface, and the high-magnification image (Fig. 1b) confirms the presence of irregularly shaped Mo-based nanoparticles with an average diameter of less than 50 nm uniformly grown on the surface of the CFC. The microstructure of Mo-based species was further investigated by TEM characterization (Fig. 1c). The nanoparticle morphology is not clear in the TEM image since the Mo-based nanoparticles were ultrasonically dispersed in ethanol during the TEM sample preparation, which caused the morphological destruction. The high-resolution TEM (HRTEM) images shown in Fig. 1d and e indicate that the

lattice fringe spacings are 2.7 Å and 6.3 Å, respectively. Interplanar spacing of 2.7 Å is consistent with the d spacing of (100) planes of hexagonal MoS₂, while, the interlayer spacing of 6.3 Å observed from the curled edge is slightly larger than the layer-to-layer spacing of 6.15 Å in bulk MoS₂. It is worth noting that many crystal fringes along the curled edge are discontinuous, which can be attributed to the existence of rich defects,²⁵ as indexed in the white dotted circular area. The defect-rich structure resulted from the excess thiourea, which can be adsorbed on the surface of primary nanocrystallites, partially hindering the oriented crystal growth and leading to the formation of a defect-rich structure with quasiperiodic configuration.²⁵ The high-angle annular dark-field scanning transmission electron microscopy (HAADF–STEM) analysis (Fig. 1f) and the corresponding elemental mapping images (Fig. 1g–i) show the presence and uniform distribution of Mo and S elements, corresponding to the obtained MoS₂. Possibly owing to the low content and poor crystallinity of the formed MoS₂ nanoparticles, only diffraction peaks at 24.7° and 43.5° of hexagonal graphite carbon (JCPDS No. 13-0148) in the XRD patterns (Fig. S4, ESI[†]) can be observed for the MoS₂/CFC.

In order to analyze the surface chemical and oxidation states of the Mo⁶⁺-adsorbed CFC and as-prepared defective MoS₂/CFC, X-ray photoelectron spectroscopy (XPS) was carried out. The peaks related to Mo in the survey spectrum (Fig. S5a, ESI[†]) indicate the adsorption of Mo⁶⁺ ions on the CFC substrate. The high resolution Mo 3d spectrum (Fig. S5b, ESI[†]) shows two characteristic peaks at the binding energies of 231.9 and 235.1 eV, which can be attributed to the orbitals of Mo^{VI} 3d_{5/2} and Mo^{VI} 3d_{3/2}, respectively. Fig. S5c (ESI[†]) shows the high resolution S 2p spectrum, confirming no S-related species in the Mo⁶⁺-adsorbed CFC. As shown in Fig. S5d (ESI[†]), the O 1s peak located at 530.0 eV corresponds to the binding energy of Mo–O bonds,³⁹ which comes from the original reagent Na₂MoO₄. The peaks located at 532.3 eV and 534.6 eV can be attributed to C–O and H–O bonds. After the VPH reaction, the signal of S 2p was observed from the survey spectrum (Fig. 2a) of the obtained MoS₂/CFC. Detailed compositional analysis (Table S1, ESI[†]) reveals that the atomic ratio of Mo:S is 1:2.94, giving the direct evidence that the existence of unsaturated sulfur atoms in the defective MoS₂/CFC. As shown in Fig. 2b, two characteristic peaks arising from Mo 3d_{5/2} and Mo 3d_{3/2} orbitals are located at 229.0 and 232.3 eV, respectively, which corresponds to the +4 oxidation state, suggesting the dominance of Mo^{IV} in the product.⁴⁰ A small S 2s peak is centered at 226.2 eV, whereas the S 2p region (Fig. 2c) exhibits primarily a single doublet with the 2p_{3/2} peak at 161.8 eV, which is consistent with the –2 oxidation state of sulfur.⁴⁰ Two peaks of 163.5 and 164.4 eV represent the exposed sulfur atoms.⁴¹ The high resolution O 1s peak located at 530.0 eV disappeared in the obtained MoS₂/CFC (Fig. 2d), which can be attributed to the breaking of Mo–O bonds. In conclusion, the XPS measurement results show that the unsaturated sulfur atoms in the defective MoS₂/CFC produce a large number of Mo vacancies.

CAL was first selected as a classical model to evaluate the selective hydrogenation performance of α,β -unsaturated aldehydes over the as-prepared defective MoS₂/CFC. As shown

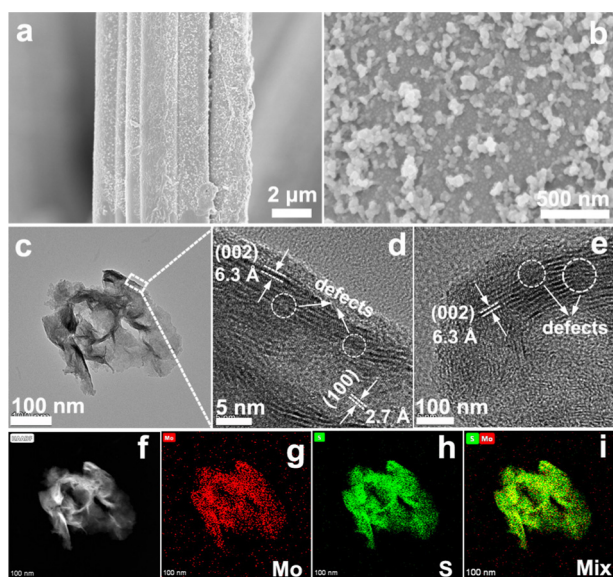


Fig. 1 MoS₂/CFC: (a) low-magnification SEM image; (b) high-magnification SEM image; (c) TEM image; (d) HRTEM image; (e) high-angle annular dark-field scanning transmission electron microscopy (HAADF–STEM) and (f–i) corresponding elemental mapping images.



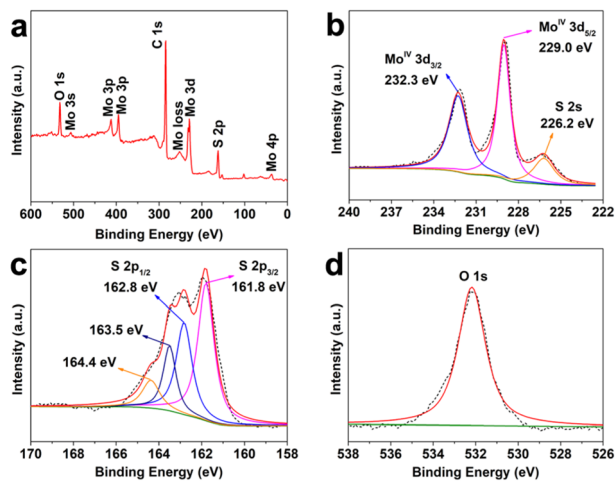


Fig. 2 (a) Surface survey XPS spectrum of MoS₂/CFC. High-resolution XPS spectra of (b) Mo 3d, (c) S 2p and (d) O 1s.

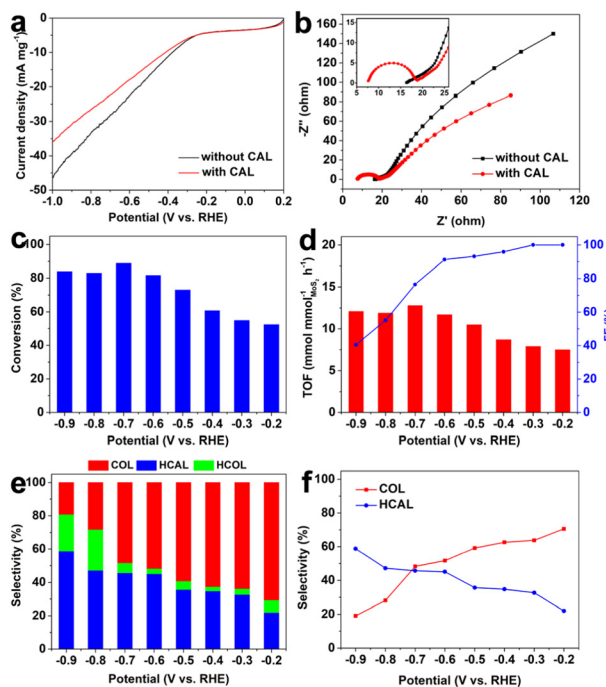


Fig. 3 (a) LSV curves and (b) EIS spectra of MoS₂/CFC in 0.1 M PBS electrolyte (pH = 7.0) with and without CAL; Potential-dependent performance of CAL hydrogenation over MoS₂/CFC: (c) conversion, (d) TOF and FE, (e) and (f) selectivity.

in the LSV curves (Fig. 3a), the addition of CAL into the electrolyte slightly decreased the current density, which was because the electrocatalytic hydrogenation of CAL was more sluggish than hydrogen evolution by water decomposition, and the occupation of active sites by CAL molecules hindered hydrogen adsorption and evolution.¹⁸ In the EIS spectra (Fig. 3b), the high-frequency intercept on the real axis represents the Ohmic series resistance (R_s), and the semicircle in the high-frequency region corresponds to the charge-transfer resistance (R_{ct}) of the electrolyte/electrode interface, which strongly influences the activity

of an electrocatalyst.³¹ Clearly, the R_s and R_{ct} values after the addition of the CAL precursor were smaller than that for pure water decomposition, indicating the higher electrocatalytic hydrogenation activity of the defective MoS₂/CFC toward the CAL precursor.

Subsequently, the potential-dependent results showed excellent CAL conversion yield, turnover frequency, faradaic efficiency and product selectivity over the defective MoS₂/CFC (Fig. 3c–f, Fig. S6, S7 and Table S2, ESI[†]). As shown in Fig. 3c, the conversion of CAL increased with the increase of applied potentials, and reached a maximum value of 88.8% at -0.7 V vs. RHE. However, when the potential became more negative, the conversion decreased, mainly due to the competitive HER process. When considering the intrinsic activity of electrocatalysts, the most important figure of merit is the turnover frequency (TOF), which is the number of moles of products generated per unit time per active site.⁴² As shown in Fig. 3d, the variation tendency of TOF was consistent with the conversion in Fig. 3c, and reached a maximum value of 12.8 mmol mmol_{MoS₂}⁻¹ h⁻¹ at -0.7 V vs. RHE. The FE values of CAL hydrogenation decreased with the increase of applied potentials (Fig. 3d), attributing to the enhanced HER activity. The detailed FE values for COL, HCAL and HCOL are shown in Fig. S7 (ESI[†]). Fig. 3e and f are the histogram and the corresponding point plot of product selectivity with voltage change, respectively. From which it can be seen clearly that, with the increase of applied potentials from -0.2 V to -0.9 V vs. RHE, the selectivity of COL gradually decreased from 70.6% to 19.1%, whereas the selectivity of HCAL gradually increased from 22.0% to 58.8%. In other words, the hydrogenation of CAL to COL was easier to achieve under lower applied potentials, whereas the hydrogenation of CAL to HCAL was easier to achieve under higher applied potentials. Meanwhile, the selectivity of HCOL also increased with the increase of applied potential, which was because the HCAL and COL were more easily transformed into fully hydrogenated HCOL under higher applied potentials. As we know, obtaining a high COL yield under low voltage is challenging because the C=O bond is thermodynamically and kinetically more difficult to be hydrogenated than the C=C bond. In this work, the high selectivity of CAL hydrogenation can be achieved by adjusting the applied potentials. Compared with other representative reports based on electrocatalytic CAL selective hydrogenation, the defective MoS₂/CFC showed an excellent performance (Table S3, ESI[†]).

To clarify the mechanism of the high selectivity of CAL hydrogenation over the defective MoS₂/CFC, we explored the adsorption of reactants and products over defective MoS₂ and graphite carbon substrate by means of DFT calculation, where bulk MoS₂ without defects acted as a reference for a better understanding of defective MoS₂. As mentioned above, excess thiourea was a prerequisite for the formation of the defect-rich structure. Therefore, we only considered the following two defective configurations: Mo vacancies and Mo/S vacancies with a stoichiometric atomic ratio of 1:2. The adsorption of hydrogen was investigated firstly. The adsorption configurations and energies of hydrogen atom over graphite carbon,



MoS₂(002) surface, and MoS₂(002) surface with Mo vacancies or Mo/S vacancies were compared. As shown in Fig. S8 (ESI†), the corresponding adsorption energies of the hydrogen atom were calculated as 147.92, 174.98, -62.40 and -70.40 kJ mol⁻¹, respectively. Clearly, the adsorption of hydrogen atom onto the (002) surface of MoS₂ with Mo vacancies or Mo/S vacancies was easier than that of graphite carbon and bulk MoS₂(002) surface without defects, which benefited the generation of H_{ads} and the proceeding of electrocatalytic hydrogenation.

Subsequently, the adsorption of a CAL molecule and its hydrogenation products HCAL and COL over the Mo-vacancy MoS₂(002) surface was explored. After optimizing the most stable adsorption configurations, a CAL molecule was adsorbed on the MoS₂(002) surface through a parallel configuration with an adsorption energy of -90.43 kJ mol⁻¹ (Fig. 4a). In detail, a CAL molecule was bound to the MoS₂(002) surface through a phenyl group, C=C bond, and C=O bond. After the C=C bond was fully hydrogenated with H_{ads}, the resulting HCAL was interacted with the MoS₂(002) surface through the phenyl group and the C=O bond with an adsorption energy of -56.14 kJ mol⁻¹ (Fig. 4b). On the other hand, when the C=O bond was fully hydrogenated with H_{ads}, the obtained COL was attracted onto the MoS₂(002) surface *via* the phenyl group, C=C bond and -OH group with an adsorption energy of -93.22 kJ mol⁻¹ (Fig. 4c). If assuming the binding energy (E_b) of the same functional group remains constant in different compounds, $E_{b,C=C}$ and $E_{b,C=O}$ can be roughly estimated from the calculated adsorption energy (ΔE_{ads}) of CAL, HCAL, COL, and POL (*n*-propanol, adsorbed mainly through -OH group, Fig. 4d) by the following equations:¹⁸

$$E_{b,C=C} = \Delta E_{ads,CAL} - \Delta E_{ads,HCAL} \quad (5)$$

$$E_{b,C=O} = \Delta E_{ads,CAL} - \Delta E_{ads,COL} + \Delta E_{ads,POL} \quad (6)$$

As listed in Table 1, the calculated binding energy of the C=O group of CAL ($E_{b,C=O} = -39.39$ kJ mol⁻¹) was more negative than that of the C=C group ($E_{b,C=C} = -34.29$ kJ mol⁻¹), suggesting the C=O group was more active than the C=C

group in CAL hydrogenation to generate COL over Mo-vacancy MoS₂.

Based on the most stable adsorption configurations of CAL, HCAL and COL on the Mo-vacancy MoS₂(002) surface, we further calculated the reaction energies for the hydrogenation of C=C (from CAL to HCAL) and C=O (from CAL to COL). As shown in Fig. 5 and Table 1, the reaction energy change from CAL to COL ($\Delta G = -83.88$ kJ mol⁻¹) was more negative than that from CAL to HCAL ($\Delta G = -75.79$ kJ mol⁻¹), indicating the hydrogenation from CAL to COL was easier than that from CAL to HCAL. In other words, during the hydrogenation of CAL over Mo-vacancy MoS₂, COL was easier to form under lower applied potentials, and HCAL was easier to form under higher applied potentials to overcome the reaction energy, according with our experimental results (Fig. 3e and f).

Besides, the adsorption energies of different reacting species, the binding energies of C=C and C=O, and the reaction energies for the hydrogenation of C=C and C=O over Mo/S vacancies MoS₂(002) surface were calculated. As shown in Fig. S9 (ESI†), Fig. 5 and Table 1, although the calculated binding energy of the C=O group of CAL ($E_{b,C=O} = -42.47$ kJ mol⁻¹) was much more negative than that of the C=C group ($E_{b,C=C} = -21.09$ kJ mol⁻¹), the reaction energy change from CAL to HCAL ($\Delta G = -88.88$ kJ mol⁻¹) was more negative than that from CAL to COL ($\Delta G = -80.12$ kJ mol⁻¹), indicating the hydrogenation from CAL to HCAL was easier than that from CAL to COL. The calculation data was not according with our experimental results, suggesting that Mo/S vacancies were not the real catalytic active centers of the defective MoS₂/CFC in this work.

Meanwhile, the adsorption energies of different reacting species, the binding energies of C=C and C=O, and the reaction energies for the hydrogenation of C=C and C=O over the bulk MoS₂(002) surface without defects were calculated. As shown in Fig. S10 (ESI†), Fig. 5 and Table 1, although the calculated binding energy of the C=O group of CAL ($E_{b,C=O} = -39.68$ kJ mol⁻¹) was more negative than that of C=C group ($E_{b,C=C} = -32.36$ kJ mol⁻¹), the reaction energy changes from CAL to COL ($\Delta G = -78.20$ kJ mol⁻¹) and from CAL to HCAL ($\Delta G = -77.62$ kJ mol⁻¹) were almost identical, suggesting no significant difference in product selectivity of CAL hydrogenation.

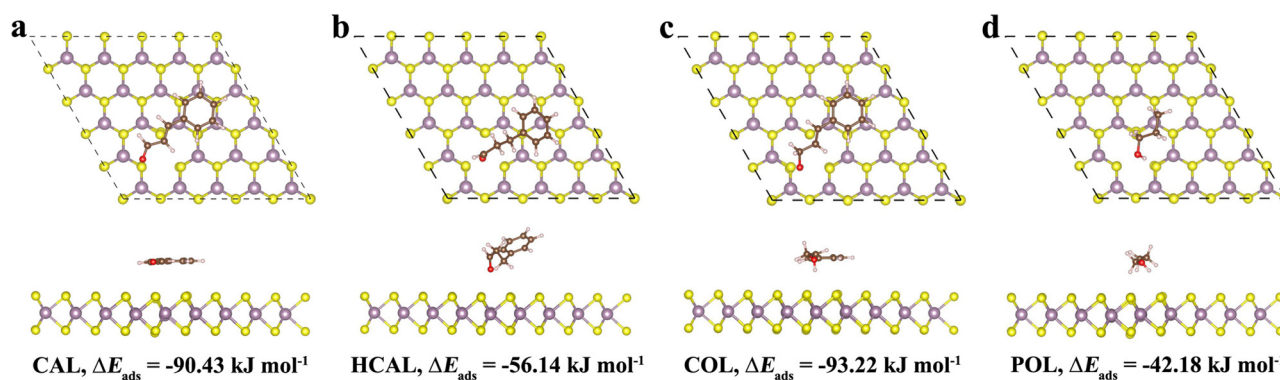


Fig. 4 Adsorption configurations and energies of different reacting species on the Mo-vacancy MoS₂(002) surface (brown sphere: C, white sphere: H, red sphere: O, yellow sphere: S, and purple sphere: Mo).



Table 1 Adsorption energy of different species, binding energy of different groups, and reaction energy of CAL hydrogenation over graphite carbon, bulk MoS₂(002) surface, MoS₂(002) surface with Mo vacancies or Mo/S vacancies

Catalyst	ΔE_{ads} (kJ mol ⁻¹)					E_{b} (kJ mol ⁻¹)		Reaction energy (kJ mol ⁻¹)	
	H	CAL	HCAL	COL	POL	C=C	C=O	CAL → AHCAL	CAL → COL
Carbon ^a	147.92	-54.02	-45.74	-82.91	-9.63	-8.38	19.26	-101.69	-104.39
MoS ₂ ^b	174.98	-90.91	-58.55	-93.60	-42.37	-32.36	-39.68	-77.62	-78.20
MoS ₂ -MoVac ^c	-62.40	-90.43	-56.14	-93.22	-42.18	-34.29	-39.39	-75.79	-83.88
MoS ₂ -MoSVac ^d	-70.40	-85.32	-64.23	-89.94	-47.09	-21.09	-42.47	-88.88	-80.12

^a Carbon: graphite carbon substrate CFC. ^b MoS₂: bulk MoS₂(002) surface without defects. ^c MoS₂-MoVac: MoS₂(002) surface with Mo vacancies.

^d MoS₂-MoSVac: MoS₂(002) surface with Mo/S vacancies.

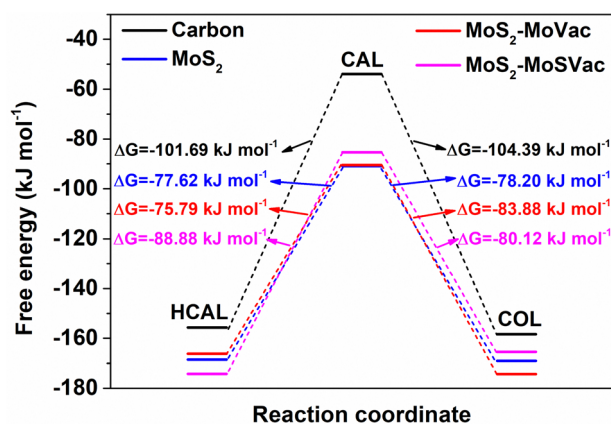


Fig. 5 Reaction pathway and energy diagram of CAL hydrogenation that proceeded over graphite carbon (denoted as carbon), bulk MoS₂(002) surface (denoted as MoS₂), Mo-vacancy MoS₂(002) surface (denoted as MoS₂-MoVac), and Mo/S-vacancy MoS₂(002) surface (denoted as MoS₂-MoSVac).

To confirm the calculation results, bulk MoS₂ without defects was synthesized using stoichiometric Mo and S sources. As shown in Fig. S11a (ESI[†]), the obtained bulk MoS₂ are nanosheets with a uniform lateral size of about 200 nm. Two peaks in the low-angle region (9.2° and 17.8°) in the XRD pattern (Fig. S11b, ESI[†]) correspond to the oxygen-incorporated MoS₂, which could be further proved by the enlarged interlayer spacing of 9.5 Å and the uniform distribution of O element (Fig. S11c-h, ESI[†]).²⁴ The XPS measurement results show that the atom ratio of Mo:S in the bulk MoS₂ is 1:2.01 (Table S1 and Fig. S12, ESI[†]). The electrocatalytic hydrogenation performance of CAL over the bulk MoS₂ was also investigated. As shown in Fig. S13 (ESI[†]), the conversion of CAL increased with the increase of applied potentials. However, the selectivity between COL and HCAL remained almost identical, which corresponded to the calculation results.

In order to explore the catalytic effect of CFC substrate in the defective MoS₂/CFC in electrocatalytic CAL hydrogenation, contrast experiments were carried out. Bare CFC without MoS₂ loading was used as an electrode directly to catalyze CAL hydrogenation under -0.7 V vs. RHE, exhibiting an 60.0% conversion of CAL and an 100.0% selectivity of HCAL. By contrast, the defective MoS₂/CFC displayed an 88.8% conversion of CAL, an 45.7% selectivity of HCAL and an 48.3% selectivity of COL under the identical conditions, respectively.

These results clearly indicated that the hydrogenation of C=O was more active on the defective MoS₂, whereas the hydrogenation of C=C was more active on graphite carbon. The calculation results (Fig. S14, ESI[†], Fig. 5 and Table 1) indicated that although the reaction energy change from CAL to COL ($\Delta G = -104.39$ kJ mol⁻¹) was more negative than that from CAL to HCAL ($\Delta G = -101.69$ kJ mol⁻¹) on graphite carbon, the calculated binding energy of the C=C group of CAL ($E_{\text{b,C=C}} = -8.38$ kJ mol⁻¹) was much more negative than that of the C=O group ($E_{\text{b,C=O}} = 19.26$ kJ mol⁻¹). Therefore, C=C was more active in CAL hydrogenation to generate HCAL.

The changes in the structure and composition of the defective MoS₂/CFC after ECH measurements were investigated to evaluate the stability of the catalyst. The XRD pattern (Fig. S15, ESI[†]) only displayed two main diffraction peaks of graphite carbon, attributed to the low content and poor crystallinity of the loaded MoS₂ nanoparticles. The SEM images (Fig. S16a and b, ESI[†]) showed no obvious changes, which can be further confirmed by TEM characterization (Fig. S16c-i, ESI[†]). Fig. S17 (ESI[†]) showed the XPS spectra of MoS₂/CFC after ECH. Compared to the initial MoS₂/CFC (Fig. 2), both Mo 3d, S 2s and S 2p peaks shifted to higher binding energies, indicating that both Mo and S gave electrons to other components during the ECH process. Here, CAL molecules captured electrons to proceed the hydrogenation reduction reaction, suggesting that Mo and S served as “electron donors”, CAL served as an “electron acceptor”, respectively, and the electrons transferred from Mo and S to CAL during the ECH process. In addition, MoS₂/CFC exhibited high stability for the electrocatalytic hydrogenation of CAL (Fig. S18, ESI[†]).

Furthermore, other typical unsaturated aldehydes such as furfural (FAL) were selected to evaluate the selective hydrogenation performance over the as-prepared defective MoS₂/CFC. As shown in Fig. S19 (ESI[†]), Fig. 6 and Table S4 (ESI[†]), furfuryl alcohol (FOL) was the only hydrogenation product in a wide potential range from -0.2 to -0.9 V vs. RHE, indicating 100.0% hydrogenation selectivity. The conversion of FAL and TOF value increased with the increase of applied potentials, and reached a maximum value of 92.6% and 13.3 mmol mmol_{MoS₂}⁻¹ h⁻¹ at -0.8 V vs. RHE. The corresponding FE values of FAL hydrogenation generally decreased with the increase of applied potentials, attributing to the enhanced HER activity.

In addition, the hydrogenation performance of typical saturated aldehyde such as benzaldehyde over the as-prepared



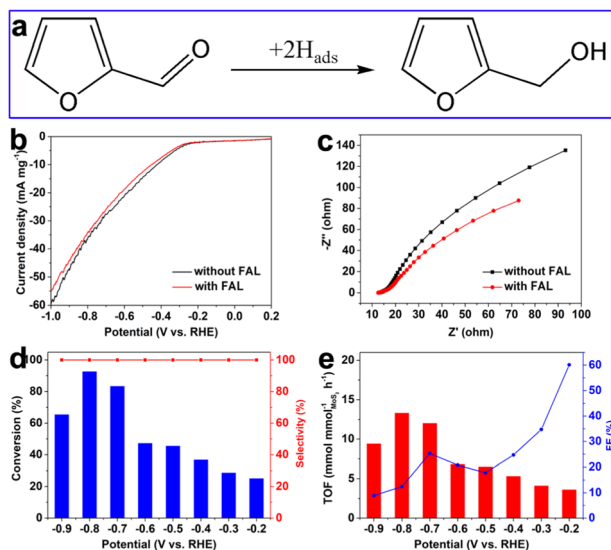


Fig. 6 (a) The hydrogenation pathways of furfural (FAL), (b) LSV curves and (c) EIS spectra of MoS₂/CFC in 0.1 M PBS electrolyte (pH = 7.0) with and without FAL; potential-dependent performance of FAL hydrogenation over MoS₂/CFC: (d) conversion and selectivity, (e) TOF and FE.

defective MoS₂/CFC was also explored. As shown in Fig. S20 (ESI[†]), Fig. 7 and Table S5 (ESI[†]), benzyl alcohol was the only hydrogenation product in a wide potential range from -0.3 to -1.0 V vs. RHE with a 100.0% hydrogenation selectivity. The conversion and TOF values increased with the increase of applied potentials, and the corresponding FE values gradually decreased with the increase of applied potentials, which was due to the enhanced competitive HER activity. These results

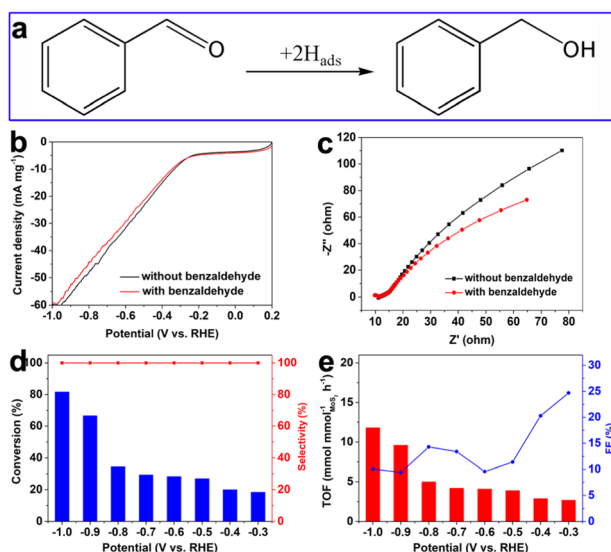


Fig. 7 (a) The hydrogenation pathways of benzaldehyde, (b) LSV curves and (c) EIS spectra of MoS₂/CFC in 0.1 M PBS electrolyte (pH = 7.0) with and without benzaldehyde; potential-dependent performance of benzaldehyde hydrogenation over MoS₂/CFC: (d) conversion and selectivity, (e) TOF and FE.

confirmed the universality of the defective MoS₂/CFC catalyst in the electrocatalytic hydrogenation reduction of aldehydes.

4. Conclusions

In summary, defective MoS₂ nanoparticles were successfully grown on commercial carbon fiber cloth (MoS₂/CFC) via a simple vapor-phase hydrothermal approach and directly used as the electrode electrocatalyst without further treatment for ECH in 0.1 M phosphate buffer electrolyte (pH = 7.0). The experimental results suggested that the as-prepared MoS₂/CFC exhibited the highest CAL conversion of 88.8% with a maximum TOF value of 12.8 mmol mmol_{MoS₂}⁻¹ h⁻¹ at -0.7 V vs. RHE. With the increase of applied potentials from -0.2 V to -0.9 V vs. RHE, the selectivity of COL gradually decreased from 70.6% to 19.1%, whereas the selectivity of HCAL gradually increased from 22.0% to 58.8%, indicating that the hydrogenation of CAL to COL was easier to achieve under a lower applied potential, whereas the hydrogenation of CAL to HCAL was easier to achieve under a higher applied potential. DFT calculations proved that the real catalytic active center was the Mo vacancies in the defective MoS₂/CFC, which facilitated the adsorption of hydrogen atoms, benefiting the generation of H_{ads} and the proceeding electrocatalytic hydrogenation. Furthermore, the ECH performance of the defective MoS₂/CFC for other unsaturated aldehyde such as FAL and saturated aldehyde such as benzaldehyde hydrogenation was also explored, exhibiting the universality in ECH of unsaturated and saturated aldehydes. This is inspirational for electrocatalyst design through defect engineering in electrocatalytic organic synthesis, especially in electrocatalytic hydrogenation of unsaturated or saturated aldehydes and ketones to produce the corresponding alcohols.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

This work was financially supported by the Key Research and Development Projects of Shaanxi Province (Grant No. 2021GY-126), the Key Science and Technology Project of Northwest Institute for Non-ferrous Metal Research (Grant No. YK2020-6), and the National Natural Science Foundation of China (Grant No. 61804154 and 12147219). We thank the LvLiang Cloud Computing Centre of China, TianHe-2 and Hefei advanced computing center for theoretical calculations and the C3S2 computing center in Huzhou University for calculation support.

Notes and references

- 1 M. C. Leech and K. Lam, A practical guide to electrosynthesis, *Nat. Rev. Chem.*, 2022, **6**, 275–286.



- 2 S. A. Akhade, N. Singh, O. Y. Gutierrez, J. Lopez-Ruiz, H. Wang, J. D. Holladay, Y. Liu, A. Karkamkar, R. S. Weber, A. B. Padmaperuma, M. S. Lee, G. A. Whyatt, M. Elliott, J. E. Holladay, J. L. Male, J. A. Lercher, R. Rousseau and V. A. Glezakou, Electrocatalytic hydrogenation of biomass-derived organics: a review, *Chem. Rev.*, 2020, **120**, 11370–11419.
- 3 J. Song, Z. F. Huang, L. Pan, K. Li, X. Zhang, L. Wang and J. J. Zou, Review on selective hydrogenation of nitroarene by catalytic, photocatalytic and electrocatalytic reactions, *Appl. Catal., B*, 2018, **227**, 386–408.
- 4 W. Xu, C. Yu, J. Chen and Z. Liu, Electrochemical hydrogenation of biomass-based furfural in aqueous media by Cu catalyst supported on N-doped hierarchically porous carbon, *Appl. Catal., B*, 2022, **305**, 121062.
- 5 P. Zhang and L. Sun, Electrocatalytic hydrogenation and oxidation in aqueous conditions, *Chin. J. Chem.*, 2020, **38**, 996–1004.
- 6 X. Chong, C. Liu, Y. Huang, C. Huang and B. Zhang, Potential-tuned selective electrosynthesis of azoxy-, azo- and amino-aromatics over a CoP nanosheet cathode, *Natl. Sci. Rev.*, 2019, **7**, 285–295.
- 7 X. H. Chadderdon, D. J. Chadderdon, J. E. Matthiesen, Y. Qiu, J. M. Carraher, J. P. Tessonier and W. Li, Mechanisms of furfural reduction on metal electrodes: distinguishing pathways for selective hydrogenation of bioderived oxygenates, *J. Am. Chem. Soc.*, 2017, **139**, 14120–14128.
- 8 K. Li and Y. Sun, Electrocatalytic upgrading of biomass-derived intermediate compounds to value-added products, *Chem. – Eur. J.*, 2018, **24**, 18258–18270.
- 9 S. Han, Y. Shi, C. Wang, C. Liu and B. Zhang, Hollow cobalt sulfide nanocapsules for electrocatalytic selective transfer hydrogenation of cinnamaldehyde with water, *Cell Rep. Phys. Sci.*, 2021, **2**, 100337.
- 10 M. Zhao, K. Yuan, Y. Wang, G. Li, J. Guo, L. Gu, W. Hu, H. Zhao and Z. Tang, Metal-organic frameworks as selectivity regulators for hydrogenation reactions, *Nature*, 2016, **539**, 76–80.
- 11 X. Wang, X. Liang, P. Geng and Q. Li, Recent advances in selective hydrogenation of cinnamaldehyde over supported metal-based catalysts, *ACS Catal.*, 2020, **10**, 2395–2412.
- 12 S. Bai, L. Bu, Q. Shao, X. Zhu and X. Huang, Multicomponent Pt-based zigzag nanowires as selectivity controllers for selective hydrogenation reactions, *J. Am. Chem. Soc.*, 2018, **140**, 8384–8387.
- 13 M. J. Torres, P. Sánchez, A. de Lucas-Consuegra and A. R. de la Osa, Electrocatalytic hydrogenation of cinnamaldehyde in a PEM cell: the role of sodium hydroxide and platinum loading, *Mol. Catal.*, 2020, **492**, 110936.
- 14 T. Wu, H. Meng and R. Dang, Amorphous Ta₂O₅-supported Ru as an efficient electrocatalyst for selective hydrogenation of cinnamaldehyde with water as the hydrogen source, *Inorg. Chem. Front.*, 2021, **8**, 4712–4719.
- 15 H. Wang, S. Bai, Y. Pi, Q. Shao, Y. Tan and X. Huang, A strongly coupled ultrasmall Pt₃Co nanoparticle-ultrathin Co(OH)₂ nanosheet architecture enhances selective hydrogenation of α,β -unsaturated aldehydes, *ACS Catal.*, 2019, **9**, 154–159.
- 16 X. Lan and T. Wang, Highly selective catalysts for the hydrogenation of unsaturated aldehydes: a review, *ACS Catal.*, 2020, **10**, 2764–2790.
- 17 X. Lan, K. Xue and T. Wang, Combined synergetic and steric effects for highly selective hydrogenation of unsaturated aldehyde, *J. Catal.*, 2019, **372**, 49–60.
- 18 X. Huang, L. Zhang, C. Li, L. Tan and Z. Wei, High selective electrochemical hydrogenation of cinnamaldehyde to cinnamyl alcohol on RuO₂-SnO₂-TiO₂/Ti electrode, *ACS Catal.*, 2019, **9**, 11307–11316.
- 19 Y. Kwon, K. J. P. Schouten, J. C. van der Waal, E. de Jong and M. T. M. Koper, Electrocatalytic conversion of furanic compounds, *ACS Catal.*, 2016, **6**, 6704–6717.
- 20 C. J. Bondue, F. Calle-Vallejo, M. C. Figueiredo and M. T. M. Koper, Structural principles to steer the selectivity of the electrocatalytic reduction of aliphatic ketones on platinum, *Nat. Catal.*, 2019, **2**, 243–250.
- 21 C. J. Bondue and M. T. M. Koper, Electrochemical reduction of the carbonyl functional group: the importance of adsorption geometry, molecular structure, and electrode surface structure, *J. Am. Chem. Soc.*, 2019, **141**, 12071–12078.
- 22 Y. Kwon, Y. Y. Birdja, S. Raoufmoghaddam and M. T. M. Koper, Electrocatalytic hydrogenation of 5-hydroxymethylfurfural in acidic solution, *ChemSusChem*, 2015, **8**, 1745–1751.
- 23 J. Tan, W. Zhang, Y. Shu, H. Lu, Y. Tang and Q. Gao, Interlayer engineering of molybdenum disulfide toward efficient electrocatalytic hydrogenation, *Sci. Bull.*, 2021, **66**, 1003–1012.
- 24 J. Xie, J. Zhang, S. Li, F. Grote, X. Zhang, H. Zhang, R. Wang, Y. Lei, B. Pan and Y. Xie, Controllable disorder engineering in oxygen-incorporated MoS₂ ultrathin nanosheets for efficient hydrogen evolution, *J. Am. Chem. Soc.*, 2013, **135**, 17881–17888.
- 25 J. Xie, H. Zhang, S. Li, R. Wang, X. Sun, M. Zhou, J. Zhou, X. W. Lou and Y. Xie, Defect-rich MoS₂ ultrathin nanosheets with additional active edge sites for enhanced electrocatalytic hydrogen evolution, *Adv. Mater.*, 2013, **25**, 5807–5813.
- 26 Y. Li, H. Wang, L. Xie, Y. Liang, G. Hong and H. Dai, MoS₂ nanoparticles grown on graphene: an advanced catalyst for the hydrogen evolution reaction, *J. Am. Chem. Soc.*, 2011, **133**, 7296–7299.
- 27 J. V. Lauritsen, J. Kibsgaard, S. Helveg, H. Topsøe, B. S. Clausen, E. Lægsgaard and F. Besenbacher, Size-dependent structure of MoS₂ nanocrystals, *Nat. Nanotechnol.*, 2007, **2**, 53–58.
- 28 A. Albu-Yaron, M. Levy, R. Tenne, R. Popovitz-Biro, M. Weidenbach, M. Bar-Sadan, L. Houben, A. N. Enyashin, G. Seifert, D. Feuermann, E. A. Katz and J. M. Gordon, MoS₂ hybrid nanostructures: from octahedral to quasi-spherical shells within individual nanoparticles, *Angew. Chem., Int. Ed.*, 2011, **50**, 1810–1814.
- 29 C. B. Roxlo, H. W. Deckman, J. Gland, S. D. Cameron and R. R. Chianelli, Edge surfaces in lithographically textured molybdenum disulfide, *Science*, 1987, **235**, 1629–1631.
- 30 T. Wu, G. Wang, X. Zhu, P. Liu, X. Zhang, H. Zhang, Y. Zhang and H. Zhao, Growth and in situ transformation



- of TiO₂ and HTiOF₃ crystals on chitosan-polyvinyl alcohol co-polymer substrates under vapor phase hydrothermal conditions, *Nano Res.*, 2016, **9**, 745–754.
- 31 T. Wu, X. Zhu, G. Wang, Y. Zhang, H. Zhang and H. Zhao, Vapor-phase hydrothermal growth of single crystalline NiS₂ nanostructure film on carbon fiber cloth for electrocatalytic oxidation of alcohols to ketones and simultaneous H₂ evolution, *Nano Res.*, 2018, **11**, 1004–1017.
- 32 T. Wu, M. Han, X. Zhu, G. Wang, Y. Zhang, H. Zhang and H. Zhao, Experimental and theoretical understanding on electrochemical activation and inactivation processes of Nb₃O₇(OH) for ambient electrosynthesis of NH₃, *J. Mater. Chem. A*, 2019, **7**, 16969–16978.
- 33 G. Kresse and J. Furthmüller, Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 34 G. Kresse and J. Furthmüller, Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 35 G. Kresse and J. Hafner, *Ab initio* molecular-dynamics simulation of the liquid-metal-amorphous-semiconductor transition in germanium, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **49**, 14251–14269.
- 36 P. E. Blöchl, Projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**, 17953–17979.
- 37 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized gradient approximation made simple, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 38 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1976, **13**, 5188–5192.
- 39 Y. Sun, X. Hu, W. Luo and Y. Huang, Self-assembled hierarchical MoO₂/graphene nanoarchitectures and their application as a high-performance anode material for lithium-ion batteries, *ACS Nano*, 2011, **5**, 7100–7107.
- 40 J. Kibsgaard, Z. Chen, B. N. Reinecke and T. F. Jaramillo, Engineering the surface structure of MoS₂ to preferentially expose active edge sites for electrocatalysis, *Nat. Mater.*, 2012, **11**, 963–969.
- 41 W. J. Xie, X. Li and F. J. Zhang, Mo-vacancy induced high performance for photocatalytic hydrogen production over MoS₂ nanosheets cocatalyst, *Chem. Phys. Lett.*, 2020, **746**, 137276.
- 42 J. Masa, C. Andronesco and W. Schuhmann, Electrocatalysis as the nexus for sustainable renewable energy: the gordian knot of activity, stability, and selectivity, *Angew. Chem., Int. Ed.*, 2020, **59**, 15298–15312.

