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## Electroreduction of carbon dioxide to formate using highly efficient bimetallic Sn–Pd aerogels<sup>†</sup>

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Electrochemical reduction of carbon dioxide ( $\text{CO}_2$ ) to valuable materials is a promising approach to suppress atmospheric  $\text{CO}_2$  levels. In order to bring this strategy to a commercial scale, the design of efficient, cost-effective, and robust catalysts is essential. Current advances in  $\text{CO}_2$  conversion technology use bimetallic components that enhance electrocatalysis *via* the introduction of binding site diversity. In this work, Sn–Pd bimetallic aerogels supported by carbon nanotubes (Sn–Pd/CNT) demonstrate selective electroreduction of  $\text{CO}_2$  to formate in ambient conditions. Amino substituents were introduced as an additional  $\text{CO}_2$  capture site (Sn–Pd/CNT–NH<sub>2</sub>), further enhancing the electrocatalytic activity and resulting in 91% formate selectively and a current density of  $-39 \text{ mA cm}^{-2}$  at  $-0.4 \text{ V}$  vs. RHE. The results demonstrate the potential of alloying Sn with other earth-abundant metals to promote the electrochemical conversion of  $\text{CO}_2$  to value-added materials. We believe this study provides valuable insights into the intricate relationship of bimetallic aerogels and shows the potential of the –NH<sub>2</sub> group as a facilitator for  $\text{CO}_2$  capture and conversion that will inspire new forays into the development of competitive catalytic systems.

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## Introduction

Increasing carbon dioxide ( $\text{CO}_2$ ) levels in the atmosphere are considered as a major contributor to climate change.<sup>1</sup> Electrocatalytic carbon dioxide reduction reactions ( $\text{CO}_2$ RRs) to value added materials such as formic acid ( $\text{HCOOH}$ ), carbon monoxide ( $\text{CO}$ ), methane ( $\text{CH}_4$ ), methanol ( $\text{CH}_3\text{OH}$ ), and ethanol ( $\text{C}_2\text{H}_5\text{OH}$ ), have drawn significant attention to mitigate climate change.<sup>2–8</sup> Formate, in particular, is valued as an ideal candidate for hydrogen storage and fuel cell processes.<sup>9,10</sup> However, due to the complexity of its multi-electron reaction pathway and competition with hydrogen evolution reactions (HERs), the reaction is sluggish and hindered by a large activation overpotential.<sup>11</sup>

There has been considerable progress on improving electrocatalytic activity for  $\text{CO}_2$  reduction by introducing binding site diversity.<sup>12</sup> Although various metals, such as Pd,<sup>13</sup> Pb,<sup>14</sup> Bi,<sup>15,16</sup> Sn,<sup>17–19</sup> Ag,<sup>20</sup> and In<sup>21</sup> themselves demonstrate high selectivity

for formate production, recent reports show that bimetallic alloys of these metals can increase catalytic activity even further.<sup>22,23</sup> Additionally, Sn-based<sup>24–27</sup> and Pd-based<sup>28,29</sup> bimetallic catalysts are superior in several aspects where monometallic catalysts are lacking: in reducing large overpotentials and improving surface stability towards  $\text{CO}_2$ RR.<sup>30–34</sup>

In addition to tailoring the elemental composition affect catalytic performance, increasing the catalytically active surface area is another strategy that can be used to enhance current density.<sup>35,36</sup> To this end, geometric factors such as metal dispersion, surface morphology, and porosity play a key role.<sup>37</sup> Aerogels stand out among similar porous structural materials in facilitating electron transport and optimizing catalytically active surface area.<sup>38–45</sup> Owing to their porous and three-dimensional (3D) structure, they can dramatically enhance mass transport and facilitate  $\text{CO}_2$ RR.<sup>46–48</sup> Liu *et al.*<sup>49</sup> have reported a Sn nanoparticle-modified 3D carbon nanotube (CNT) aerogel whose unique porous, hierarchical structure was instrumental in increasing current density and effectively enhancing the selectivity of formate production. These studies stress the import of morphology control on the performance of Sn-based catalysts for the  $\text{CO}_2$ RR.<sup>50</sup>

Knowing that Pd surface is poisoned by  $\text{CO}$ ,<sup>51</sup> in this project, we took an advantage of Sn surface to enhance the adsorption of  $\text{CO}_2$ , but weakened the  $\text{CO}$  binding on Pd, and have designed a series of bimetallic Sn–Pd aerogels, to improve their stability and overall catalytic performance through the synergy between

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Pd and Sn surface for electroreduction of  $\text{CO}_2$ . Aerogels of these bimetallic systems were employed to maximize the surface area of the catalytic layer through the creation of a highly porous nanostructure.<sup>45,48</sup> Building on previous works, carbon nanotubes (CNTs) were incorporated throughout the aerogel to enhance the electron conductivity of the structure (Sn-Pd/CNT).<sup>52</sup> Amine groups have been shown to have a high capability of capturing  $\text{CO}_2$ ,<sup>53–55</sup> and are integrated into our design to serve as an additional  $\text{CO}_2$  capture site and localized proton source (Sn-Pd/CNT-NH<sub>2</sub>). Using this design, we could successfully reduce  $\text{CO}_2$  to formate with 91% selectivity and current density of  $-39 \text{ mA cm}^{-2}$  at  $-0.4 \text{ V}$  vs. RHE. Comparative studies of these bimetallic systems with their monometallic Sn and Pd aerogels counterparts will be discussed in this work.

## Experimental

A series of 3D Pd, Sn-Pd, Sn-Pd-CNT, and Sn-Pd-CNT-NH<sub>2</sub> aerogels were synthesized through a one-pot and surfactant-free technique by reducing  $\text{H}_2\text{PdCl}_4$  and  $\text{SnCl}_2$  with glyoxylic acid using lyophilization (ESI† Part 3).<sup>48</sup> The spongy, porous morphology characteristic of aerogels was visualized using Scanning Electron Microscopy (SEM) (Fig. 1a, and Fig. S1, ESI†). The uniform distribution of atoms within the aerogel's structure affirms a homogeneous distribution of corresponding Pd, Sn, N, and C components (Fig. 1b–e). Transmission electron microscopy (TEM) images highlight the structural presence of CNTs in the case of the Sn-Pd/CNT-NH<sub>2</sub> aerogel (Fig. S2, ESI†).

X-Ray photoelectron spectroscopy (XPS) gives insight into the chemical environments of the bimetallic material, where the two characteristic peaks with the binding energy of 335 eV and 340 eV correspond to Pd 3d<sub>3/2</sub> and Pd 3d<sub>5/2</sub> and peaks at 486.1 and 492.5 eV are attributed to Sn 3d<sub>5/2</sub> and Sn 3d<sub>3/2</sub>, respectively (Fig. 2a, b, and Fig. S3–S9, ESI†). In the presence of CNTs, a slight upward shift in the binding energy could be indicative of a highly dispersive and synergetic relationship

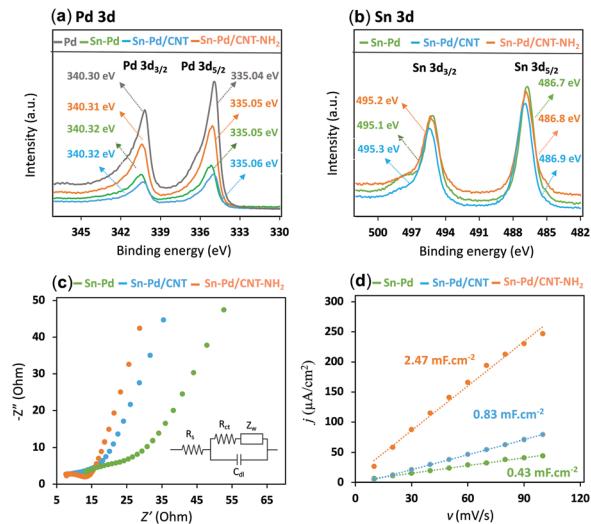


Fig. 2 X-ray photoelectron spectroscopy (XPS) spectra survey comparison of: (a) Pd 3d spectra of Pd, Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> aerogels; (b) Sn 3d spectra of Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> aerogels. (c) Electrochemical impedance spectroscopy (EIS) of Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> in saturated solution with  $\text{CO}_2$  and 0.1 M  $\text{KHCO}_3$  with amplitude of 5 mV in a frequency range from 10 kHz to 0.1 Hz. (d) Current density difference between cathodic and anodic sweeps at  $-0.44 \text{ V}$  vs. scan rates in 2.5 mM  $[\text{Fe}(\text{CN})_6]^{3-/-4-}$  and 0.2 M  $\text{KNO}_3$ .

between the two materials.<sup>56,57</sup> The Sn-Pd/CNT-NH<sub>2</sub> aerogel, was characterized further by X-ray diffraction (XRD) (Fig. S10, ESI†). Peaks seen at  $40.1^\circ$ ,  $46.6^\circ$ ,  $68.1^\circ$ , and  $82.1^\circ$  are characteristic diffraction peaks of Pd crystals; whereas, peaks at  $26.5^\circ$  and  $34.4^\circ$  belong to CNT<sup>58</sup> and Sn which confirm the successful formation of the aerogel. These results are in agreement with previous reports.<sup>25,59,60</sup>

The nature of the charge-transfer resistance in the presence and absence of CNTs was investigated using electrochemical impedance spectroscopy (EIS) (Fig. 2c). The semicircular arcs at the high-frequency regions of the Nyquist plots are characteristic of a capacitance double layer between the electrolyte and the electrode interface; which reflects the charge transfer resistance of an electrochemical reaction ( $R_{\text{ct}}$ ).<sup>61</sup> The values of  $R_{\text{ct}}$  from equivalent circuit fitting for Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> are an estimated 23.1, 17.9, and 8.7  $\Omega$ , respectively. The lower  $R_{\text{ct}}$  in Sn-Pd/CNT and Sn-Pd/CNT-NH<sub>2</sub> compared to Sn-Pd underscore the significant improvement to charge transfer and electrical conductivity the CNT and CNT-NH<sub>2</sub> offer.<sup>62,63</sup>

The electronic structure of Sn-Pd-based catalysts has been reported previously. The bimetallic Pd-Sn alloys have a unique electronic structure similar to that of noble metals that forms when rehybridization of the Sn sp-orbitals and Pd d-orbitals occurs.<sup>64–66</sup> In the case of Sn-Pd/CNT and Sn-Pd/CNT-NH<sub>2</sub>, the  $\pi$ -orbitals of CNTs can weakly bond with Sn-Pd through hybridization between the carbon  $\pi$ -orbitals and the high energy d-orbitals of the metals;<sup>67,68</sup> strengthening the electronic structure and thereby facilitating charge transfer. To further examine the charge transfer of the synthesized aerogels, the

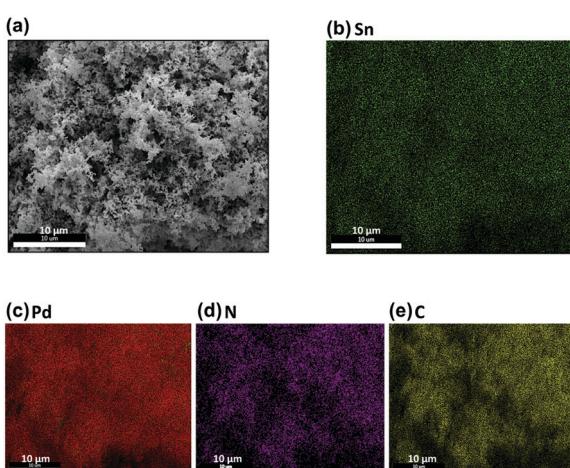


Fig. 1 (a) Scanning Electron Microscopy (SEM) of Sn-Pd-CNT-NH<sub>2</sub> aerogel with a scale bar of 10  $\mu\text{m}$ . (b–e) Atom distribution of Sn, Pd, N, and C, respectively.

standard exchange current density ( $i_0$ ) was calculated using the following equation (eqn (1)): <sup>69</sup>

$$i_0 = \frac{RT}{nFR_{ct}} \quad (1)$$

where  $R$ ,  $T$ ,  $n$ ,  $F$ , and  $R_{ct}$  are the gas constant, temperature, number of electrons involved in the reaction, faradaic constant, and the calculated charge transfer resistance obtained from EIS. The  $i_0$  values for Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> aerogels were calculated as  $1.1 \times 10^{-3}$ ,  $1.5 \times 10^{-3}$ , and  $3.0 \times 10^{-3}$ , respectively, highlighting the superior charge transfer of the Sn-Pd/CNT-NH<sub>2</sub> aerogel.<sup>70</sup>

To shed light on the influence of CNTs and -NH<sub>2</sub> on surface area activity, it is important to quantify the electrochemically active surface area (ECSA) of the modified electrode surfaces as described in ESI† Part 4 (Fig. S11, ESI†).<sup>71</sup> The  $C_{dl}$  was estimated by plotting the  $j$  ( $\mu\text{A cm}^{-2}$ ) at  $-0.44$  V vs. Ag/AgCl against the scan rate ( $\text{mV s}^{-1}$ ) and it is twice the slopes of linear fit (Fig. 2d). The roughness factor ( $R_f$ ) of the electrodes can be calculated using eqn (2):<sup>72,73</sup>

$$R_f = \frac{C_{dl}}{60} \quad (2)$$

where  $C_{dl}$  is the double-layer capacitive current (more detail in ESI†) and the  $C_{dl}$  of an ideal smooth surface is  $60 \mu\text{F cm}^{-2}$ . From eqn (2) and the electrode surface area of  $0.071 \text{ cm}^2$ , the ECSA was calculated using eqn (3):

$$\text{ECSA} = R_f \times S_{\text{electrode}} \quad (3)$$

Combining with the CVs results in Fig. S11 (ESI†) and actual catalyst load on glassy carbon electrode, the ECSA for Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> was calculated as  $0.25$ ,  $0.49$ , and  $1.46 \text{ cm}^2$ , respectively. The higher ECSA for the Sn-Pd/CNT-NH<sub>2</sub> indicates its larger catalytic activity compared to Sn-Pd and Sn-Pd/CNT. The higher ESCA of Sn-Pd/CNT compared to Sn-Pd could be due to the greater charge transfer that CNT facilitates between the aerogel and the electrode surface.<sup>74–77</sup> Further improvement to the ESCA is seen in the case of Sn-Pd/CNT-NH<sub>2</sub>, which can be due to enhanced electron–proton transfer of the -NH<sub>2</sub> groups.<sup>57</sup>

## Results and discussion

To evaluate the electroactivity of the synthesized complexes, electrochemical reduction of CO<sub>2</sub> was studied with linear sweep voltammetry (LSV) in an aqueous  $0.1 \text{ M KHCO}_3$  solution. A two-compartment electrochemical H-cell comprised of a three-electrode system with immobilized aerogels on a glassy carbon working electrode (GCE), platinum counter electrode, and a silver/silver chloride (Ag/AgCl) reference electrode.<sup>78</sup> To prepare the working electrodes,  $10 \mu\text{mol}$  of the synthesized Pd-Sn aerogels were immobilized onto a  $0.3 \text{ cm}$  diameter GCE with a catalyst loading concentration of  $1.57 \times 10^{-7} \text{ mol cm}^{-2}$  relative to the geometrical electrode area ( $0.071 \text{ cm}^2$ ).<sup>55</sup> To incorporate the CNTs, Sn-Pd and CNTs were dissolved in a

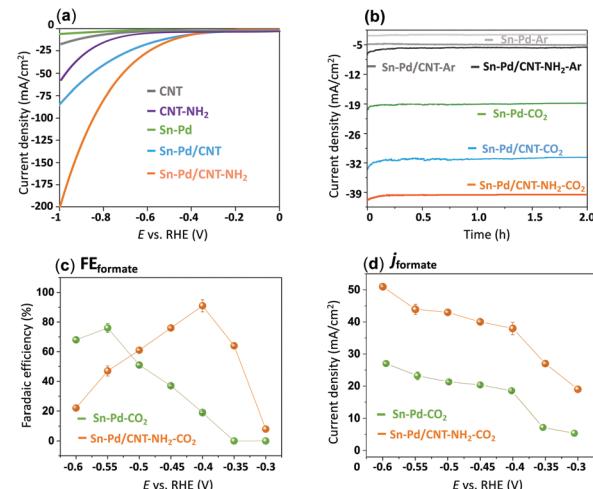


Fig. 3 (a) Linear sweep voltammetry (LSV) comparison of bare CNT, bare CNT-NH<sub>2</sub>, Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> in  $0.1 \text{ M KHCO}_3$ ; (b) chronoamperometry comparison of Sn-Pd, Sn-Pd/CNT, Sn-Pd/CNT-NH<sub>2</sub> in  $0.1 \text{ M KHCO}_3$  at  $-0.4 \text{ V}$  vs. RHE. (c) Faradaic efficiency (FE<sub>formate</sub>), and (d) current density ( $j_{\text{formate}}$ ) comparison of Sn-Pd, and Sn-Pd/CNT-NH<sub>2</sub> at  $-0.30$ ,  $-0.35$ ,  $-0.4$ ,  $-0.45$ ,  $-0.5$ ,  $-0.55$  and  $-0.6 \text{ V}$  vs. RHE in  $0.1 \text{ M KHCO}_3$ .

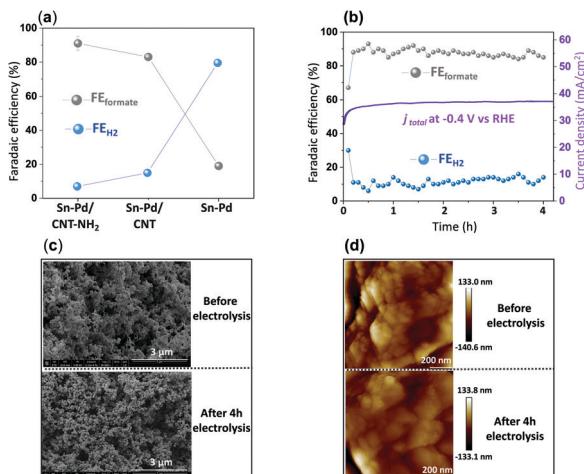
DMF solution and sonicated for 20 minutes to form a homogeneous suspension of Sn-Pd/CNT and Sn-Pd/CNT-NH<sub>2</sub>.<sup>48</sup>

Fig. 3a compares the LSV of bare CNT and CNT-NH<sub>2</sub> with the immobilized aerogels Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> onto GCE in  $0.1 \text{ M KHCO}_3$  in the presence of CO<sub>2</sub>. Exposure of the aerogels to CO<sub>2</sub> resulted in a dramatic increase to the current density beginning at  $-0.67$ ,  $-0.42$ , and  $-0.38$ , V vs. RHE for compounds Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub>, respectively, which could be due to either electroreduction of CO<sub>2</sub> or HER (Fig. S12, ESI†). Comparing pristine CNT with CNT-NH<sub>2</sub> in Fig. 3a finds a higher current density and lower potential in the presence of -NH<sub>2</sub>. This observation aligns with what has been reported previously.<sup>54</sup> Achieving the highest catalytic activity with the lowest overpotential in the case of Sn-Pd/CNT-NH<sub>2</sub> further highlights the role of -NH<sub>2</sub> in capturing CO<sub>2</sub>.

The proficiency of the bimetallic Sn-Pds towards CO<sub>2</sub> electroreduction was evaluated under chronoamperometric conditions. The experiments were conducted at  $-0.3$ ,  $-0.35$ ,  $-0.4$ ,  $-0.45$ ,  $-0.5$ ,  $-0.55$  and  $-0.6 \text{ V}$  vs. RHE in  $0.1 \text{ M KHCO}_3$  for  $2 \text{ h}$  under sealed conditions with vigorous magnetic stirring. Formate and H<sub>2</sub> were detected as the sole products of the electrochemical CO<sub>2</sub>RR after gas chromatography (GC) sampling of the reaction headspace and <sup>1</sup>H NMR and HPLC analysis of the liquid solution. As shown in Fig. S14–S16 (ESI†), the product distribution is strongly dependent on the applied potential. A greater decrease in FE<sub>formate</sub> was seen at higher overpotentials in competition with an uptick in HERs.<sup>54</sup>

The optimal overpotential for Sn-Pd was found to be  $-0.55 \text{ V}$  vs. RHE, and much lower optimal potential of  $-0.4 \text{ V}$  vs. RHE was observed in the case of Sn-Pd-CNT and Sn-Pd/CNT-NH<sub>2</sub> (Fig. 3b). This could be due to the support of the CNT





**Fig. 4** (a) Faradaic efficiency (FE) and current density ( $j$ ) comparison of Sn-Pd, Sn-Pd/CNT, and Sn-Pd/CNT-NH<sub>2</sub> at  $-0.4$  V vs. RHE in  $0.1$  M KHCO<sub>3</sub>; (b) long-term stability studies of Sn-Pd/CNT-NH<sub>2</sub> for CO<sub>2</sub> electroreduction operated in  $0.1$  M KHCO<sub>3</sub> at  $-0.4$  V vs. RHE for  $4$  h; (c) SEM, and (d) AFM comparison of Sn-Pd/CNT-NH<sub>2</sub> before and after  $4$  hours electrochemical CO<sub>2</sub>RR.

scaffold, and  $-\text{NH}_2$  group which works to both improve conductivity and CO<sub>2</sub> capture throughout the catalytic layer. Introducing amino groups, such as in Pd/CNT-NH<sub>2</sub>, served dual purposes as a localized proton source and as an additional CO<sub>2</sub> capture site, facilitate even higher catalytic activity and selectivity. To confirm these phenomena and important role of CNT and  $-\text{NH}_2$  in the CO<sub>2</sub>RR, the FE<sub>formate</sub> and  $j$ <sub>formate</sub> of Sn-Pd with Sn-Pd/CNT-NH<sub>2</sub> was compared in Fig. 3c and d at the potentials of  $-0.3$  V to  $-0.6$  vs. RHE in  $0.1$  M KHCO<sub>3</sub>. It clearly indicates both product selectivity and catalytic activity have been enhanced  $\sim 2$ -folds at the optimal potential of  $-0.4$  V vs. RHE in the case of Sn-Pd/CNT-NH<sub>2</sub>.

Fig. 4a demonstrates the current densities and selectivities of both Sn-Pd/CNT (FE<sub>formate</sub>: 83%,  $j$ <sub>formate</sub>:  $-29.6$  mA cm<sup>-2</sup>), and Sn-Pd/CNT-NH<sub>2</sub> (FE<sub>formate</sub>: 91%,  $j$ <sub>formate</sub>:  $-39.1$  mA cm<sup>-2</sup>) at  $-0.4$  V vs. RHE were significantly higher than that of Sn-Pd (FE<sub>formate</sub>: 19%,  $j$ <sub>formate</sub>:  $-18.2$  mA cm<sup>-2</sup>). The long-term, operational stability of Sn-Pd/CNT-NH<sub>2</sub> was monitored continuously for  $4$  h in KHCO<sub>3</sub> at  $-0.4$  V vs. RHE (Fig. 4b). The current density ( $-39$  mA cm<sup>-2</sup>) was maintained for the entire duration and FE<sub>formate</sub> remained constant over 90%. No significant decrease in current density and product selectivity was observed during this period SEM (Fig. 4c), atomic force microscopy (AFM) (Fig. 4d), and XPS (Fig. S17, ESI†) found no significant alterations in surface morphology before and after long-term electrolysis. This phenomenon was further examined and discussed using FTIR and described in detail in the ESI† (Fig. S18).

The ability of the Sn-Pd/CNT-NH<sub>2</sub> aerogel to capture and adsorb CO<sub>2</sub> was measured at varying temperatures based on the weight change of the aerogel before and after the introduction of CO<sub>2</sub>.<sup>79</sup> Sn-Pd/CNT-NH<sub>2</sub> (12 mg) was heated to  $110$  °C in a sealed chamber under N<sub>2</sub> with a flow rate of  $40$  sccm to remove any air or existing moisture. After applying CO<sub>2</sub>, the weight

change of the sample during adsorption and desorption process was calculated every  $2$  min at  $25$  °C under CO<sub>2</sub> with a flow rate of  $40$  sccm. The CO<sub>2</sub> uptake per mole of Sn-Pd/CNT-NH<sub>2</sub> slowly increased over time, with a maximum uptake of  $6.89$  mmol g<sup>-1</sup> over  $48$  minutes (Fig. S19a, ESI†). Next, the adsorption of the aerogel was measured at several temperatures under CO<sub>2</sub> atmosphere (Fig. S19b, ESI†). CO<sub>2</sub> uptake improved with increasing temperatures, reaching a maximum of  $7.82$  mmol g<sup>-1</sup> at  $75$  °C. The stability of the aerogel was examined over several adsorption/desorption cycles at  $75$  °C and no dramatic change in the level of adsorption was observed.

To obtain insight on the CO<sub>2</sub> electroreduction reaction mechanism, Tafel slopes were calculated and performed in Fig. S20 (ESI†). The slopes were measured to be  $97$ ,  $84$ , and  $82$  mV dec<sup>-1</sup>, for Sn-Pd, Sn-Pd/CNT and Sn-Pd/CNT-NH<sub>2</sub>, respectively. Achieving smaller Tafel slope values in the case of Sn-Pd/CNT-NH<sub>2</sub>, suggest that less driving force (overpotential) is required to improve the reaction rate,<sup>80,81</sup> which is also in agreement with previous reports on the bimetallic Sn-Pd alloy which shown to be highly dependent on the catalysts' surface electronic properties.<sup>82</sup> The achieved Tafel values are close to the theoretical value for a rapid one-electron transfer step followed by a rate-determining step (RDS).<sup>25</sup> Reducing CO<sub>2</sub> to formate at such low overpotential,<sup>25,83</sup> suggest that \*OCHO is the key intermediate for the CO<sub>2</sub>RR electroreduction.<sup>84</sup> The Sn's ideal \*OCHO binding energy supports this phenomenon. These results are also consistent with previous reports on observing \*OCHO on Sn-based electrodes using spectroscopic techniques.<sup>85,86</sup>

In summary, using porous, bimetallic Sn-Pd-based aerogels paired with CNT-NH<sub>2</sub>, we could significantly increase their catalytic efficiency and product selectivity compared to their monometallic analogues, highlighting the importance of synergistic binding site diversity between Sn and Pd (Table S1, ESI†). To the best of our knowledge, this is one of the lowest overpotentials that has been reported for bimetallic compounds for CO<sub>2</sub>RR applications (Table S2, ESI†).<sup>30,34,82,87–98</sup> The essential role of the  $-\text{NH}_2$  group in acting as an additional CO<sub>2</sub> capture site and as an extra proton source played a significant role in increasing the catalytic activity and selectivity of this system.<sup>99,100</sup> The remarkable catalytic performance of the Sn-Pd aerogels reported herein make it a promising candidate for CO<sub>2</sub>RR on a large scale.

## Conclusions

In conclusion, we have synthesized and characterized a new class of Pd-Sn aerogels outfitted with CNTs and  $-\text{NH}_2$  groups. In this work, the introduction of binding site diversity increased the current density  $\sim 36$ -fold higher than its monometallic constituents. The exceptional catalytic performance achieved with Sn-Pd/CNT-NH<sub>2</sub> demonstrates high selectivity (FE<sub>formate</sub>: 91%), current density ( $j$ <sub>formate</sub>:  $-39.1$  mA cm<sup>-2</sup>), and stability at low overpotential of  $0.4$  V. The potential for

bimetallic tunability and the unique 3D structure of aerogels facilitated greater CO<sub>2</sub> capture and conversion and makes them an attractive class of electrocatalysts for electroreduction of CO<sub>2</sub> to value-added materials.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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