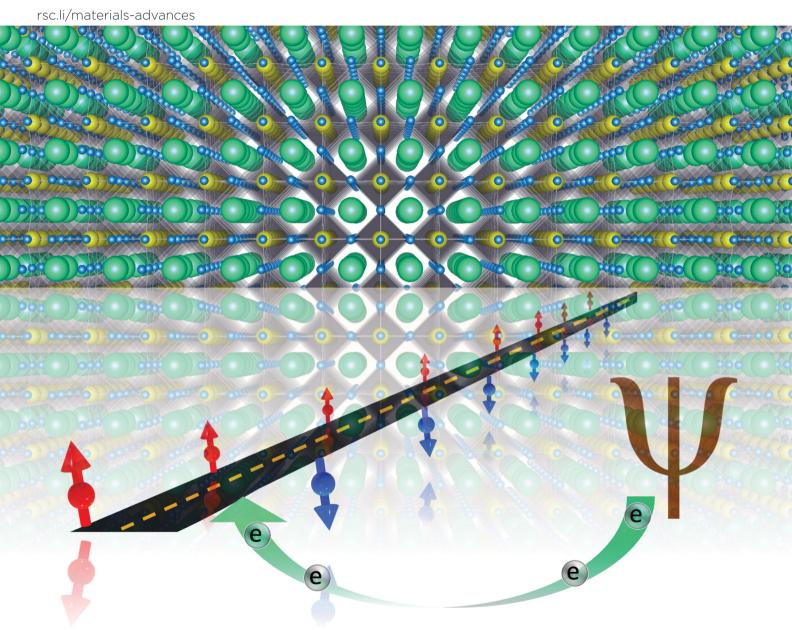
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Rashba spin splitting and anomalous spin textures in the bulk ferroelectric oxide perovskite KIO3†

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The momentum-dependent Rashba and Dresselhaus spin splitting has gained much attention for its highly promising applications in spintronics. A non-centrosymmetric structure and the presence of spinorbit coupling (SOC) lead to the momentum-dependent spin splitting of degenerate bands at non-timereversal-invariant k-points. This lifts the Kramers degeneracy leading to Rashba and Dresselhaus splitting. In the search for new ferroelectric Rashba semiconductors, here we present ferroelectric oxide perovskite KIO3, where the presence of a heavy element (I), contributing to significant SOC and inversion asymmetric nature induces interesting band splitting. By employing the state-of-the-art density functional theory (DFT) with the semi-local and hybrid functional (HSE06) combined with SOC, we find non-negligible spin splitting effects at the conduction band minimum (CBm) and valence band maximum (VBM) for R3m and R3c phases. For a deeper understanding of the observed spin splitting, we have analyzed the spin textures within the combined framework of DFT and the $k \cdot p$ model Hamiltonian. Linear Rashba terms successfully explain splitting at the VBM. However, cubic terms become important in realizing spin-orientation near the CBm. In the R3c phase, the four-band k·p model Hamiltonian is needed to completely understand the anomalous nature of the spin textures, which is beyond the conventional linear Rashba and Dresselhaus splitting. Our results show the enhancement in Rashba parameters on tuning the epitaxial strain. Furthermore, we have observed a reversal of spin-orientation upon switching the direction of polarization.

1 Introduction

Ferroelectric Rashba semiconductors have recently created a huge sensation in the field of spintronics owing to their robust spontaneous electrical polarization. 1-6 These materials find applications in spin field effect transistors, ferroelectric tunnel junctions, and storage and memory devices. 7-12 The long-range order dipoles aligned in the same direction induce polarization in ferroelectric materials, leading to inversion asymmetry. Ferroelectric Rashba semiconductors interlink the phenomena of Rashba-type splitting and ferroelectricity, enabling the electric control of electron spin. In ferroelectric Rashba semiconductors, Rashba parameters can be enhanced via the application of ferroelectric (FE) polarization.

Ferroelectric materials, because of their robust spontaneous electrical polarization, are widely used in various applications. Interestingly, the spin-orientation can be inverted by reversing the direction of polarization using an external electric field.¹³

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Electrical control of the spin degrees of freedom makes them suitable for spintronic devices. Some well known examples of ferroelectric Rashba semiconductors are KTaO₃, 14 BiAlO₃, 13 LiZnSb¹⁵ and FASnI₃ (FA = formamidinium). ¹⁶ GeTe was the first ferroelectric material, in which the Rashba effect was predicted theoretically 17 and observed experimentally. 18 However, it has a tendency to form Ge vacancies, 19 which leads to p-type semiconducting behavior. This in turn poses a challenge to electric control of the spin. In this context, ferroelectric oxide perovskites such as KIO₃ (KIO) show excellent piezoelectric, pyroelectric and non-linear optical properties. 20,21 KIO, in particular, is experimentally synthesized at high temperature in a non-centrosymmetric rhombohedral phase with R3m space group symmetry.22 A thermodynamically favorable rhombohedral R3c phase was also theoretically predicted. 21,23,24 In both phases, the distortion of the octahedra centered at the I-atom induces spontaneous FE polarization. Furthermore, the presence of a heavy element (I), contributing to significant spin-orbit coupling (SOC) and inversion asymmetric nature may induce interesting Rashba- and Dresselhaus-type band splitting.

Note that SOC and broken inversion symmetry play a pivotal role for the materials to exhibit Rashba and Dresselhaus

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effects.7, 12,25,26 In crystals lacking inversion symmetry, a relativistically moving electron experiences a Lorentz-transformed magnetic field due to a finite potential gradient. This results in spin-based splitting of degenerate bands at non-time-reversalinvariant k-points, which lifts the Kramers degeneracy leading to Rashba and Dresselhaus splitting. The spin-orientation is determined by the momentum-dependent spin-orbit field. For acentric non-polar crystals. Dresselhaus was the first to show band splitting, which shows a cubic dependence on momentum space for zinc blende-type crystal structures.²⁷ For gyrotropic point group symmetries, linear Dresselhaus-type spin splitting can also be realized. In polar crystals and 2D electron gas, linear splitting terms are allowed as shown by Rashba and Bychkov. ^{28–30} The SOC Hamiltonian $H_{SO} = \Omega(k) \cdot \sigma$ describes these effects, where σ is the Pauli matrix vector and $\Omega(k)$ is the spin-orbit field. The latter is odd in momentum space (i.e. $\Omega(-k) = -\Omega(k)$) to preserve the time-reversal symmetry of H_{SO} . $\Omega(k)$ depends on the spatial symmetry of the system. For the simplest case, C_{2v} point group symmetry $\Omega(k)$ can be written as a vector sum of linear Rashba $(\Omega_R = \alpha_R(k_y, -k_x, 0))$ and Dresselhaus ($\Omega_D = \alpha_D(k_y, k_x, 0)$) spin-orbit fields.³¹ Here, α_R and α_D are the Rashba and Dresselhaus coefficients, respectively. These coefficients mainly depend on the amount of SOC and the symmetry of the crystal.³² Energy eigenvalues corresponding to the linear Rashba Hamiltonian are given by $E_{+}(k)$ = $\hbar^2 k^2/2m \pm \alpha_R k$. Rashba and Dresselhaus effects lead to the same type of band splitting. However, the type of splitting can be characterized by projection of the spin-orientation in Fourier space, usually referred to as the spin texture.33

In this article, we have studied the Rashba and Dresselhaus effects in R3m and R3c phases of FE oxide KIO using state-ofthe-art density functional theory (DFT) and the symmetry adapted two-band $k \cdot p$ model Hamiltonian. Firstly, we have determined the FE polarization in both the phases. Subsequently, the crucial effect of SOC has been shown in the band structures. The Rashba spin splitting energy and offset momentum have been determined from the splitting at the valence band maximum (VBM) and conduction band minimum (CBm). Furthermore, the type of splitting has been characterized by plotting the spin texture. The Rashba and Dresselhaus parameters are determined after fitting the $k \cdot p$ model Hamiltonian to the DFT band structure. Finally, the effect of polarization on the aforementioned parameters has been investigated.

2 Methodology

The calculations are performed using the Vienna ab initio simulation package (VASP)34,35 within the framework of DFT using projector augmented wave (PAW)36 pseudopotentials. The Perdew-Burke-Ernzerhof (PBE)³⁷ exchange-correlation (ε_{xc}) functional is used for DFT calculations. For better accuracy of excited state properties and validation of PBE results, the non-local Heyd-Scuseria-Ernzerhof (HSE06)³⁸ ε_{xc} functional is used. For an effective interpretation of results, the conventional hexagonal setting is also considered. A cutoff energy of 600 eV

is used throughout the calculations. Both rhombohedral phases, R3c and R3m are fully relaxed (the lattice parameters as well as atomic coordinates) using the PBE ε_{xc} functional without including SOC with $9 \times 9 \times 4$ and $9 \times 9 \times 8$ k-grids, respectively, generated using the Monkhorst-Pack scheme.³⁹ The insignificant role of SOC in relaxation is also verified by test calculations. The partial density of states (PDOS) and band structure calculations are done using a 12 \times 12 \times 6 k-grid. In structural optimization, the total energy difference between two ionic relaxation steps is set to be smaller than 10⁻⁵ eV and the tolerance on forces between two consecutive steps is set to 0.001 eV \mathring{A}^{-1} . The FE properties are calculated within the framework of Berry phase theory for polarization. 40-42 We have conducted symmetry analysis using ref. 43, Findsym, 44 Bilbao crystallographic server⁴⁵ and SEEK-PATH.⁴⁶ Finally, we have used Mathematica⁴⁷ and PyProcar⁴⁸ to plot spin textures and to solve the $k \cdot p$ model Hamiltonian. Spin textures are calculated using the expectation values of spin operators S_i , given by

$$s_i = \frac{1}{2} \langle \Psi_k | \sigma_i | \Psi_k \rangle \tag{1}$$

where σ_i are the Pauli matrices and Ψ_k is the spinor eigenfunction obtained from noncollinear spin calculations. Spin texture calculations are performed using a closely spaced 15 \times 15 k-grid around a high symmetry point (HSP) within the $k_x - k_y$ plane. The square region considered in the $k_x - k_y$ plane is bound by $|k_x| \leq 0.2 \text{ Å}^{-1}$ and $|k_y| \leq 0.2 \text{ Å}^{-1}$. We have computed the $k \cdot p$ model band structures by minimizing the summation

$$S = \sum_{i=1}^{2} \sum_{k_x, k_y, k_z} f(k_x, k_y, k_z) | \text{Det}[H(k_x, k_y, k_z) - E^i(k_x, k_y, k_z)] |^2$$
(2)

where $f(k_x, k_y, k_z)$ is the weight attached to (k_x, k_y, k_z) point. We have used a normal distribution for $f(k_x, k_y, k_z)$ centered at the (k_x, k_y, k_z) point to get a better fit near a HSP as used in ref. 49 and 50. The range of k_x and k_y used for fitting the DFT band structures is -0.125 to 0.125 Å⁻¹. The coordinate convention used for calculating the band structures, spin textures and building the Hamiltonians is explained in Section I of the ESI.† For introducing the strain, the structural parameter c is varied with respect to equilibrium parameter c_0 and the strain (ε) is calculated as

$$\varepsilon = \frac{c - c_0}{c_0} \times 100\% \tag{3}$$

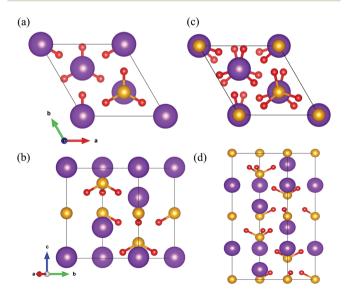
After introducing the respective strain, we have relaxed the atomic coordinates. We have varied the strain from -5% to +5%, where "-" and "+" represent the compressive and tensile strains, respectively.

3 Results and discussion

3.1 Structural and ferroelectric properties

KIO mainly exists in Pm3m, R3m and R3c space group symmetries. The $Pm\bar{3}m$ phase is centrosymmetric and contains an inversion center and therefore, does not show the Rashba-type **Materials Advances**

splitting. Hence, we have studied the non-centrosymmetric rhombohedral phases R3m and R3c in detail for Rashba and Dresselhaus properties. The R3m phase is symmorphic as all generating symmetry operations leave one common point fixed, apart from the lattice translations. In contrast, R3c phase is non-symmorphic containing the glide planes. Fig. 1 shows the conventional hexagonal unit cells of the R3m and R3c phases of KIO. The lattice parameters of both phases are given in Table 1. The calculated lattice parameters of the R3m phase are well in agreement with experimental values (a = 6.34 Å, c = 7.94 Å).²² The lattice parameters of R3c phase coincide with the crystal structure already reported using DFT (a = 6.37 Å, c = 15.92 Å).²³ At equilibrium (without considering the strain), the formation energy of the R3c phase is the lowest. Our calculated formation energy of the R3c phase is 18 meV f.u.⁻¹ lower than that for the R3m phase. For the R3m phase, we have calculated the change in the ferroelectric polarization i.e., dipole moment per unit volume $\left(\frac{p}{V}\right)$ with respect to the centrosymmetric structure, of 41 $\mu C \text{ cm}^{-2}$ along the [0001] direction in a hexagonal setting (along the [111] direction in a rhombohedral setting). However, in the R3c phase, we have found a slightly smaller polarization of 29 μ C cm⁻² along the [0001] direction in a hexagonal setting. In previous studies on BiFeO₃ thin films, it has been shown that FE polarization can be enhanced to be as large as 150 μ C cm⁻² using the strain. 51-53 In view of this, we have also verified the enhancement in FE polarization of KIO on the application of strain (the calculated lattice parameters for different strained states are given in Section II of the ESI†). The FE polarization increases (decreases) under the application of tensile (compressive) strain and direction remains the same. The FE polarization has varied from 21 to 35 μC cm⁻² and 29 to 50 μC cm⁻² in



R3m and R3c phases, respectively, on applying strain in the

range of -5% to +5%.

Fig. 1 Crystal structure of KIO₃ in the (a) and (b) R3m phase and (c) and (d) R3c phase. The violet, orange and red balls denote the K, I and O ions, respectively

Table 1 Lattice parameters and polarization for rhombohedral phases of KIO calculated using the PBE ε_{xc} functional

	Space group	a (Å)	c (Å)	$V(\mathring{A}^3)$	$P (\mu C cm^2)$
R3c 6.37 15.91 558.6 29	R3c	6.37	15.91	558.6	29
R3m 6.29 8.11 278.1 41	R3m	6.29	8.11	278.1	41

3.2 Electronic properties and spin splitting effects

Fig. 2a shows the calculated band structure of the R3m phase without SOC along the high symmetry path in the first Brillouin zone (see the Fig. 2a inset for the high symmetry path). A direct band gap of 2.51 eV is observed at the k-point A. Since the Perdew-Burke-Ernzerhof (PBE)³⁷ ε_{xc} functional is known to underestimate the band gap, it is also calculated using a nonlocal HSE06³⁸ ε_{xc} functional. A larger direct band gap of 3.50 eV is observed at the k-point A. The uppermost valence band has a width of about 2.5 eV and the electronic states are mainly derived from O-2p orbitals (see the PDOS in Fig. 2). The lowest conduction band has a width of nearly 4 eV and the electronic states are mainly derived from equal contributions of I-5p and O-2p orbitals. Fig. 2b shows the calculated band structure and PDOS with the inclusion of SOC. The VBM and CBm shift from the k-point A towards L, which is known as the offset momentum (δk) . The energy difference between the k-point A and extremum is known as the Rashba spin splitting energy (δE).

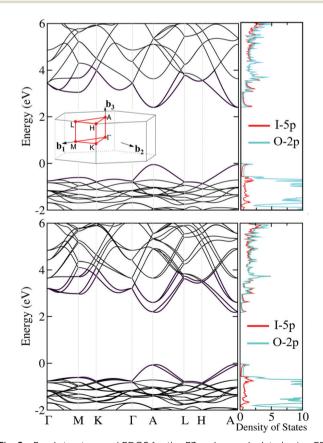


Fig. 2 Band structure and PDOS for the R3m phase calculated using PBE (a) without SOC and (b) with SOC. The Fermi energy is set to the VBM. Here, the inset shows the first Brillouin zone for the hexagonal phase.

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A slightly indirect band gap of 2.24 and 3.27 eV is observed using PBE + SOC and HSE06 + SOC, respectively. Despite the underestimation of the band gap by PBE, the band dispersion around the HSP is known to be similar to HSE06.17 We have compared the band structures obtained using PBE + SOC and HSE06 + SOC and found that they are resulting in similar Rashba parameters (see Section III of the ESI†). Therefore, all the calculations are performed using PBE, since it is more cost effective. The presence of a large SOC is attributed to the heavy elements like I. A dominant spin splitting can be seen in the plane $k_z = \frac{\pi}{2}$, which is perpendicular to the polarization axis (see Fig. 3a and b). In contrast, splitting is completely absent in the direction Γ -A, which is parallel to the polarization axis [0001]. It is consistent with the Rashba model, where splitting occurs in the direction perpendicular to the polarization axis. The δE for CB and VB is 27.3 and 18.1 meV, respectively. The δk for CB and VB is 0.047 and 0.054 Å⁻¹, respectively. Within the linear approximation, these values lead to $\alpha_R \left(\frac{2\delta E}{\delta k}\right)$ of 1.16 and 0.67 eV Å, respectively (see Section IV of the ESI†). We have observed a comparatively larger Rashba splitting at the CBm

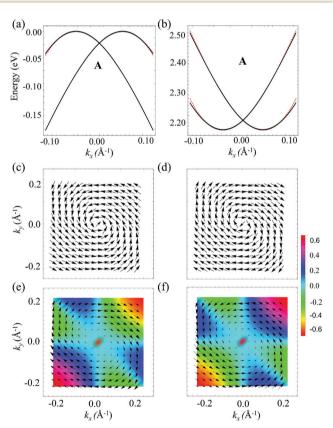


Fig. 3 (a) Valence bands and (b) conduction bands of KIO for the R3m phase with SOC along the A-L line (energy $vs. k_x$). Here black solid lines and red dashed lines are obtained by DFT and parametrization of the model, respectively. (c) and (d) Spin textures of two upper most valence bands and (e) and (f) two lowest conduction bands. The arrows and color projection represent the in-plane and out-of-plane components of the spin texture with respect to the $(k_x - k_y)$ plane.

than at the VBM due to the higher contribution of I-5p orbitals at the CBm (see Fig. 2).

In order to have a better understanding of the spin splitting nature, the spin texture is plotted near the VBM and CBm around the k-point A. The 2D spin texture is calculated by projecting the expectation values of σ_x , σ_y and σ_z in the Fourier plane (k_x-k_y) . Fig. 3[(c), (d)] and [(e), (f)] show the calculated x-, y- and zcomponents of the spin texture near the VBM and CBm, respectively. The spin textures obtained using PBE and HSE06 are in close agreement (see Fig. S4 in the ESI†). Thus, we have further used PBE for computing spin textures. The in-plane spin components (S_x) S_{ν}) show a helical nature with inner and outer bands having opposite orientations. This confirms the existence of Rashba-type splitting. A significant out of plane spin component can also be seen for the CBm, which is absent for the VBM. The out of plane spin component (S_z) has three-fold symmetry, which is in agreement with the threefold rotation symmetry of the crystal. The spin splitting and textures can be explained by two-band effective Hamiltonian including only the spin degree of freedom. The little group of the k-point A is C_{3v} , consisting of three-fold rotations C_3 , one reflection through vertical plane containing the z-axis (σ_{xz}) and two reflections through diagonal planes $(M_{d1}, M_{d2})^{54,55}$ (besides trivial identity operation). The band dispersion relation and spin texture around the k-point A in the plane orthogonal to the polar axis can be derived using all the symmetry-allowed terms such that $O^{\dagger}H(\mathbf{k})O = H(\mathbf{k})$, where O is the symmetry operation belonging to the little group. 13 The constructed two-band $k \cdot p$ Hamiltonian including linear and cubic Rashba terms satisfying the C_{3v} symmetry near the k-point A takes the form⁵⁶ (for more details see Section IV in the ESI†):

$$H_A(\mathbf{k}) = H_o(\mathbf{k}) + H_{SO} \tag{4}$$

where,

$$H_{SO} = \alpha \sigma_y k_x + \beta \sigma_x k_y + \gamma \sigma_z [(k_x^3 + k_y^3) - 3(k_x k_y^2 + k_y k_x^2)]$$
(5)

and $H_o(k)$ is the free particle Hamiltonian. α , β are the coefficients of linear terms and γ is the coefficient of the cubic term in the SOC Hamiltonian. The two energy eigenvalues of Hamiltonian are

$$E(\mathbf{k})^{\pm} = \frac{\hbar^2 k_x^2}{2m_x} + \frac{\hbar^2 k_y^2}{2m_y} \pm E_{SO}$$
 (6)

where, m_x and m_y represent the effective masses in the x and y directions, respectively. E_{SO} is the energy eigenvalue of the SOC Hamiltonian given by $E_{SO}(\mathbf{k}) = \sqrt{\alpha^2 k_x^2 + \beta^2 k_y^2 + \gamma^2 f^2(k_x, k_y)}$, where $f(k_x, k_y) = (k_x^3 + k_y^3) - 3(k_x k_y^2 + k_y k_x^2)$. Normalized spinor wavefunctions corresponding to energy eigenvalues are given by

$$\Psi_{k}^{\pm} = \frac{e^{ik \cdot r}}{\sqrt{2\pi(\rho_{\pm}^{2} + 1)}} \left(\frac{i\alpha k_{x} - \beta k_{y}}{\gamma f(k_{x}, k_{y}) \mp E_{SO}} \right)$$
(7)

where ${\rho_{\pm}}^2 = \frac{\alpha^2 k_x^2 + \beta^2 k_y^2}{\left(\gamma f\left(k_x, k_y\right) \mp E_{SO}\right)^2}$. The expectation values of spin operators are given by

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$$\{s_x, s_y, s_z\}^{\pm} = \pm \frac{1}{E_{xo}} \{\beta k_y, \alpha k_x, \gamma f(k_x, k_y)\}$$
 (8)

In-plane spin components are reproduced using α and β , whereas γ reproduces the out of plane spin component. For a specific k-point, the in-plane spin component is small as compared to out-of-plane if $\gamma f(k_x, k_y)$ is much larger than αk_x and βk_v , and vice versa. Spin textures calculated using DFT satisfy the model Hamiltonian predictions. Three-fold degeneracy of the z-component of the spin is the consequence of the cubic nature of $f(k_x, k_y)$ (see Section III in the ESI† for more details).

Fig. 3a and b show the comparison between the DFT and $k \cdot p$ model predicted band structures calculated using eqn (4) near the VBM and CBm, respectively, in the vicinity of the k-point A. The $k \cdot p$ model produces a band structure which is in close agreement with the DFT band structure. Near the k-point A, cubic terms have smaller energy contribution as compared to the linear terms in band structure calculations. In the range $|k_i| \leq 0.125 \text{ Å}^{-1}$ (i = x, y), the contribution coming from the cubic term is within 0.01 times the contribution coming from linear terms (note that the ratio of different contributions is a function of k). The energy eigenvalues of the Hamiltonian are

given by
$$E(\mathbf{k})^{\pm} = \frac{\hbar^2 k_x^2}{2m_x} + \frac{\hbar^2 k_y^2}{2m_y} \pm \sqrt{\alpha^2 k_x^2 + \beta^2 k_y^2 + \gamma^2 f^2(k_x, k_y)}$$
,

which estimate only the magnitude of α , β and γ . The signs of α , β and γ are determined by the orientation of spins in Fourier space. Rashba and Dresselhaus coefficients are defined as

 $\alpha_R = \frac{\alpha - \beta}{2}$ and $\alpha_D = \frac{\alpha + \beta}{2}$, respectively (the details can be seen in Section IV of ESI†). For the CB, α = 1.12 eV Å, β = -1.12 eV Å and $\gamma = -21.70$ eV Å³ reproduce the DFT band structure. These parameters lead to $\alpha_R = 1.12$ eV Å and $\alpha_D = 0$ eV Å, confirming that the splitting is coming mainly from the linear Rashba effect. For the VB, $\alpha = 0.63$ eV Å, $\beta = -0.63$ eV Å and $\gamma = -1.10$ eV Å³ reproduce the DFT band structure leading to purely Rashba with $\alpha_R = 0.63$ eV Å apart from the negligible cubic contribution. The Rashba parameters calculated using the $k \cdot p$ model are listed in Table 2 and are fairly close to the DFT predictions.

Fig. 4a and b show the band structures and PDOS for the R3c phase without and with inclusion of SOC, respectively. It has a direct band gap of 2.86 eV at the k-point Γ without including SOC, which is 0.41 eV larger than for the R3m phase. The contribution of atomic orbitals in PDOS near the VBM and CBm for the R3c phase is nearly similar to the R3m phase. With the inclusion of SOC, the VBM and CBm shift towards the k-point K. A slightly indirect band gap of 2.65 eV on including SOC

Table 2 Rashba parameters for band-splitting at the k-point A for R3m phase (note that $\beta = -\alpha$ in each case)

Position	m^* (m _e)	$\alpha \; (eV \; \mathring{A})$	$\gamma \; \big(eV \; \mathring{A}^3 \big)$	$\delta E \text{ (meV)}$	$\delta k (\mathring{\text{A}}^{-1})$	α_R (eV Å)
VBM	-0.72	0.63	-01.10	18.10	0.054	0.67
CBm	0.34	1.12	-21.70	27.30	0.047	1.16

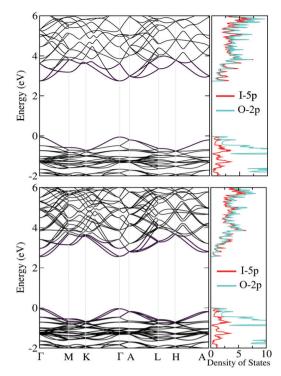


Fig. 4 Band structure and PDOS for R3c phase calculated using PBE (a) without SOC and (b) with SOC. The Fermi energy is set to the VBM.

confirms the importance of SOC in the calculations. A larger band gap is calculated to be 3.80 eV using HSE06 + SOC. Spin splitting can be seen throughout the Brillouin zone except for Γ -A, which again confirms the polarization direction parallel to Γ -A (see Fig. 4b). A dominant spin splitting can be seen along the path Γ -M, Γ -K, A-L and A-H (see Fig. 4 and 5(a)-(d)). Fig. 5(e)-(h) show the plotted spin textures near the CBm and VBM, respectively, around the k-point Γ . Fig. 5(i)–(l) show the plotted spin textures for the lowest four conduction bands, around the k-point A. Near the k-point Γ , the in-plane spin textures form a helical-type spin texture with different orientations for inner and outer bands, confirming the Rashba-type splitting of degenerate levels. A significant out of plane spin component contribution can also be seen near the CBm, which is absent near the VBM. Near the k-point A, the spin splitting leads to novel spin textures, which are beyond the superposition of linear Rashba and Dresselhaus terms. These spin textures are completely different from those obtained using the two-band $k \cdot p$ model. These spin textures are a result of coupling between the orbital and spin degrees of freedom. 49,57 The splitting near the k-point Γ can be explained by the twoband effective Hamiltonian, whereas the four-band Hamiltonian becomes important to understand the splitting near the k-point A. The little group of k-point Γ and A is the same, which is C_{3v} . The model Hamiltonian given by eqn (4) can explain the band properties of the VBM and CBm near the k-point Γ . Fig. 5a and b show that DFT and $k \cdot p$ model predicted band structures near the k-point Γ are comparable. The Rashba spin splitting energy of 6.3 and 16.5 meV is obtained for VB and CB,

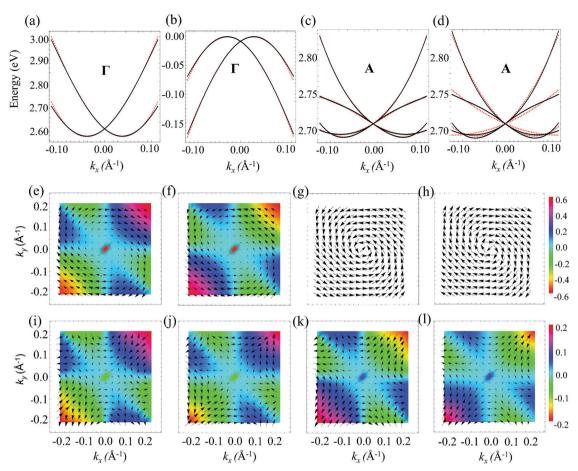


Fig. 5 (a) Valence bands and (b) conduction bands of KIO for the R3c phase with SOC along the $\Gamma-M$ line (energy $vs.\ k_x$). (c) The lowest four conduction bands around the k-point A with the full model are given in eqn (9) and (d) with the modified model excluding the terms containing the orbital degrees of freedom (obtained using $\eta = \xi = \Delta = \Lambda = K = \rho = 0$ in eqn (9)). Here, black solid lines and red dashed lines are obtained by DFT and parametrization of the models using eqn (2), respectively. Spin textures of (e) and (f) two lowest conduction bands, (g) and (h) two uppermost valence bands around k-point Γ and (i)—(l) lowest four conduction bands around the k-point Λ . The arrows and color projection represent the in-plane and out of plane components of the spin texture with respect to the $(k_x - k_y)$ plane. Our spin textures are prepared using the convention [(i)-(l)] as examples that if at the (k_x, k_y) point, energy eigenvalues associated with (i)-(l) are e_{1k} , e_{2k} , e_{3k} , e_{4k} , respectively, then $e_{1k} \le e_{2k} \le e_{3k} \le e_{4k}$.

respectively. The δk for CB and VB is 0.042 and 0.031 Å⁻¹, respectively. From DFT, the value of α_R is found to be 0.79 and 0.41 eV Å for CB and VB, respectively. For CB, fitted parameters of H_{SO} are α =0.78 eV Å, β = -0.78 eV Å and γ = -13.3 eV ų. These parameters lead to α_R = 0.78 eV Å and α_D = 0 eV Å. The corresponding parameters for VB are estimated as α = 0.40 eV Å, β = -0.40 eV Å and γ = -0.55 eV ų leading to a purely linear Rashba effect with α_R = 0.40 eV Å with negligible cubic contribution. The Rashba coefficients calculated using the $k \cdot p$ model are fairly close to the DFT predictions and shown in

Table 3 Rashba parameters for band-splitting at the k-point Γ for the R3c phase

Position	<i>m</i> * (m _e)	$\alpha \; (eV \; \mathring{A})$	$\gamma \; \big(eV \; \mathring{A}^3 \big)$	$\delta E \text{ (meV)}$	$\delta k (\mathring{\mathrm{A}}^{-1})$	α_R (eV Å)
VBM CBm	$-0.59 \\ 0.41$		$-0.55 \\ -13.27$	6.30 16.50		0.41 0.79

Table 3. For splitting near k-point A, we have used the four-band model derived using method of invariants satisfying the C_{3v} point group symmetry⁴⁹ (see Section V of the ESI†):

$$H_{A}(k) = \alpha (k_{x}^{2} + k_{y}^{2}) + \beta (k_{x}^{2} + k_{y}^{2})^{2} + \delta (k_{x}^{2} + k_{y}^{2})^{3}$$

$$+ \eta [(k_{x}^{2} - k_{y}^{2})\gamma_{z} + 2k_{x}k_{y}\gamma_{x}] + [\Delta + \Lambda (k_{x}^{2} + k_{y}^{2}) + K(k_{x}^{2} + k_{y}^{2})^{2}]\gamma_{y}\sigma_{z}$$

$$+ [\kappa + \zeta (k_{x}^{2} + k_{y}^{2})](k_{x}\sigma_{y} - k_{y}\sigma_{x}) + \lambda k_{x}(k_{x}^{2} - 3k_{x}^{2})\sigma_{z}$$

$$+ \xi k_{x}(k_{x}^{2} - 3k_{y}^{2})\gamma_{z} + \rho [(k_{x}\gamma_{x} - k_{y}\gamma_{z})\sigma_{x} - ((k_{y}\gamma_{x} - k_{x}\gamma_{z})\sigma_{y})]$$
(9)

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(a) (c) (c) (d) (d)

Fig. 6 (a) Band gap (E_g) , (b) Rashba spin splitting energy (δE) and (c) Rashba coefficients (α_R) as a function of strain (ε) . The A^1 and A^2 show the information about splitting around the k-point A in the R3m phase around the CBm and VBM, respectively. In the same manner, Γ^1 and Γ^2 show splitting in the R3c phase around the k-point Γ .

where both γ_i and σ_i are the Pauli matrices, denoting the orbital and spin degrees of freedom, respectively. Here, $\gamma_i \sigma_i$ represents the Kronecker delta product of matrices γ_i and σ_i , γ_i and σ_i denote $\gamma_i \sigma_1$ and $\gamma_1 \sigma_i$, where γ_1 and σ_1 are the identity matrices of dimension 2×2 . Fig. 5c shows the comparison between DFT and model predicted band structures for the lowest four conduction bands around the k-point A. Fig. 5d shows the DFT (same as Fig. 5c) and the modified model of eqn (9) excluding the terms containing orbital degrees of freedom predicted band structures (see Section V in the ESI†). The modified model band structure largely deviates from the DFT band structures. This confirms the significance of orbital degrees of freedom and four-band $k \cdot p$ Hamiltonian over the usual two-band $k \cdot p$ Hamiltonian to completely understand the spin splitting near the k-point A. We have observed the linear splitting of ~ 0.64 and 0.45 eV Å for the bands around the k-point A (calculated using the expression $E = E_0 \pm \lambda \delta k + b(\delta k)^2$). The CBm and VBM are in the vicinity of the k-point A, which can be brought to the k-point A using the strain for possible applications of these novel spin textures. Rashba coefficients of some selected ferroelectric materials are compared with KIO in Section VI of the ESI.†

The experimental observations show that the electronic structure, ferroelectricity, and spin splitting in oxide perovskites are tunable using the strain.⁵⁸ The polarization and Rashba splitting are sensitive to the uniaxial strain in oxide perovskites such as BiAlO₃, ¹³ KTaO₃ ⁵⁹ and PbTiO₃. ⁶⁰ In view of this, we have studied the strain dependence of δE , δk and α_R in both the phases of KIO. We have considered the effect of strain within the (0001) plane on spin splitting and energy levels for both the phases around the VBM and CBm. Such a strain can be obtained by growing KIO films on top of hexagonal (0001) or cubic (111) substrates.⁶¹ The polarization direction remains along the [0001] direction throughout the full range of strain. Fig. 6 shows the change in the band gaps, δE and α_R on application of strain. Band gap increases (decreases) under tensile (compressive) strain for R3m and R3c phases. δE decreases (increases) using the tensile (compressive) strain for conduction and valence bands. A linear trend is observed for spin splitting at the CBm and VBM in both phases. The amplitude of spin splitting can be tuned from 0.8 to 1.2 eV Å for the CBm in the R3m phase. It should be noted here that a more detailed analysis would be necessary to see if other states can be stabilized under the uniaxial strain. This can be done by calculating the total formation energy of structures with different space groups as a function of strain to know which one is the most stable at a particular strain.⁶²

The spin degeneracy can be restored when the structure is brought back to the parent centrosymmetric structure. The parent space group of R3m is $R\bar{3}m$, which is generated by the symmetry elements of R3m followed by an inversion. The polarization, P_z along the c-direction breaks the inversion symmetry. Thus, the $R\bar{3}m$ phase is centrosymmetric and band splitting is absent. It is well known that in ferroelectric Rashba semiconductors, switching the direction of polarization leads to the reversal of the spin texture. The full reversal of the spin texture on reversing the direction of polarization is also observed in KIO. To switch the direction of polarization, transition state $R\bar{3}m$ has to be crossed, which defines the potential barrier. The parent centrosymmetric structure is generated using the program PSEUDO.63 The calculated potential barrier to switch the polarization through the $R\bar{3}m$ phase is 135 meV f.u.⁻¹ for R3m. Similarly, the potential barrier for the R3c phase is found to be 112 meV f.u.⁻¹ The calculated potential barrier is comparable to 110 meV f.u.⁻¹ of LaWN₃⁴⁹ and is smaller than 130 meV f.u.⁻¹ of LiNbO₃.⁶⁴ The Rashba parameters are larger for KIO than BAO, despite a significantly larger ferroelectric polarization in BAO. Therefore, we infer that larger ferroelectric polarization does not directly imply the larger splitting. A thorough analysis of the symmetry and electronic structure is always required.

4 Conclusions

In summary, we have performed relativistic first-principles density functional theory calculations to study Rashba and Dresselhaus effects in ferroelectric rhombohedral phases of KIO with R3m and R3c space group symmetries. Ferroelectric and electronic properties are explored using DFT, which are also supported by the symmetry adapted $k \cdot p$ Hamiltonian. Near the VBM and CBm, states are mainly derived from I-5p and O-2p orbitals. A sufficiently wide and slightly indirect band gap is calculated for the R3m (2.24 (PBE + SOC), 3.27 eV (HSE06 + SOC)) and R3c (2.65 (PBE + SOC), 3.80 eV (HSE06 + SOC))

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phases. Due to a significant amount of SOC, spin splitting effects are observed at both the VBM and CBm around k-points A and Γ for R3m and R3c phases, respectively. The helical-type in-plane spin texture confirms that the spin splitting mainly consists of Rashba-type splitting. The out of plane spin texture shows importance of cubic terms in the model Hamiltonian. The two-band $k \cdot p$ Hamiltonian satisfying C_{3v} symmetry reproduces the band structure and spin texture near the CBm and VBM, that are well in agreement with the DFT results. In the R3c phase, the inclusion of four bands in the Hamiltonian incorporating orbital degrees of freedom becomes important for a complete analysis of spin splitting. Novel spin textures observed in the R3c phase are anomalous in the sense that they differ from usual spin textures and cannot be treated as linear Rashba, linear Dresselhaus or some unique combination of both. The largest Rashba coefficient is found for the CBm in the R3m phase. Furthermore, we have investigated the effect of strain on Rashba and Dresselhaus parameters and found that they increase linearly with strain. Control of spin-based properties using the external electric field makes it suitable for spintronics applications. The larger Rashba coefficient in comparison to that of other contemporary materials (viz. BAO and LaWN₃) makes KIO a promising addition into this class of materials having Rashba-based applications.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 L. W. Martin and A. M. Rappe, Nat. Rev. Mater., 2016, 2,
- 2 Y. Ma, Y. Dai, N. Yin, T. Jing and B. Huang, J. Mater. Chem. C, 2014, 2, 8539-8545.
- 3 B. Zhou, Nanoscale, 2020, 12, 5533-5542.
- 4 X. Yang, X.-M. Li, Y. Li, Y. Li, R. Sun, J.-N. Liu, X. Bai, N. Li, Z.-K. Xie, L. Su, Z.-Z. Gong, X.-Q. Zhang, W. He and Z. Cheng, Nano Lett., 2021, 21, 77-83.
- 5 S.-D. Guo, W.-Q. Mu, Y.-T. Zhu, R.-Y. Han and W.-C. Ren, J. Mater. Chem. C, 2021, 9, 2464-2473.
- 6 X.-k. Hu, Z.-x. Pang, C.-w. Zhang, P.-j. Wang, P. Li and W.-x. Ji, J. Mater. Chem. C, 2019, 7, 9406-9412.

- 7 C. Liu, H. Gao, Y. Li, K. Wang, L. A. Burton and W. Ren, J. Mater. Chem. C, 2020, 8, 5143-5149.
- 8 V. Garcia and M. Bibes, Nat. Commun., 2014, 5, 1-12.
- 9 H. Wang, P. Gopal, S. Picozzi, S. Curtarolo, M. B. Nardelli and J. Sławińska, npj Comput. Mater., 2020, 6, 1-7.
- 10 B. T. Zhou, K. Taguchi, Y. Kawaguchi, Y. Tanaka and K. T. Law, Commun. Phys., 2019, 2, 1-7.
- 11 M. Kawano, Y. Onose and C. Hotta, Commun. Phys., 2019, 2, 1-8.
- 12 S.-D. Guo, Y.-T. Zhu, W.-Q. Mu and X.-Q. Chen, J. Mater. Chem. C, 2021, 9, 7465-7473.
- 13 L. G. D. da Silveira, P. Barone and S. Picozzi, Phys. Rev. B, 2016, 93, 245159.
- 14 P. King, R. He, T. Eknapakul, P. Buaphet, S.-K. Mo, Y. Kaneko, S. Harashima, Y. Hikita, M. Bahramy and C. Bell, et al., Phys. Rev. Lett., 2012, 108, 117602.
- 15 A. Narayan, Phys. Rev. B: Condens. Matter Mater. Phys., 2015, 92, 220101.
- 16 A. Stroppa, D. Di Sante, P. Barone, M. Bokdam, G. Kresse, C. Franchini, M.-H. Whangbo and S. Picozzi, Nat. Commun., 2014, 5, 1-8.
- 17 D. Di Sante, P. Barone, R. Bertacco and S. Picozzi, Adv. Mater., 2013, 25, 509-513.
- 18 M. Liebmann, C. Rinaldi, D. Di Sante, J. Kellner, C. Pauly, R. N. Wang, J. E. Boschker, A. Giussani, S. Bertoli and M. Cantoni, et al., Adv. Mater., 2016, 28, 560-565.
- 19 A. V. Kolobov, J. Tominaga, P. Fons and T. Uruga, Appl. Phys. Lett., 2003, 82, 382-384.
- 20 X. Yin and M. K. Lü, Appl. Phys. Lett., 1992, 60, 2849-2850.
- 21 M. A. Kader, F. El-Kabbany, H. Naguib and W. Gamal, J. Phys.: Conf. Ser., 2013, 012036.
- 22 H. Kasatani, S. Aoyagi, Y. Kuroiwa, K. Yagi, R. Katayama and H. Terauchi, Nucl. Instrum. Methods Phys. Res., Sect. B, 2003, **199**, 49-53.
- 23 L. Bayarjargal, L. Wiehl, A. Friedrich, B. Winkler, E. A. Juarez-Arellano, W. Morgenroth and E. Haussuehl, J. Phys.: Condens. Matter, 2012, 24, 325401.
- 24 G. Crane, Z. Kristallogr. Cryst. Mater., 1975, 141, 312-313.
- 25 Y.-z. Jia, W.-x. Ji, C.-w. Zhang, P. Li, M.-j. Ren and P.-j. Wang, J. Mater. Chem. C, 2016, 4, 8750-8757.
- 26 S.-s. Li, W.-x. Ji, C.-w. Zhang, P. Li and P.-j. Wang, J. Mater. Chem. C, 2016, 4, 2243-2251.
- 27 G. Dresselhaus, Phys. Rev., 1955, 100, 580-586.
- 28 E. I. Rashba, Phys. Solid State, 1960, 2, 1109-1122.
- 29 F. T. Vas'ko, ZhETF Pisma Redaktsiiu, 1979, 30, 574.
- 30 Y. A. Bychkov and E. I. Rashba, J. Phys. C: Solid State Phys., 1984, 17, 6039-6045.
- 31 L. Tao and E. Y. Tsymbal, J. Phys. D: Appl. Phys., 2021, 54, 113001.
- 32 L. Tao, T. R. Paudel, A. A. Kovalev and E. Y. Tsymbal, Phys. Rev. B, 2017, 95, 245141.
- 33 R. Winkler, S. Papadakis, E. De Poortere and M. Shayegan, Spin-Orbit Coupling in Two-Dimensional Electron and Hole Systems, Springer, 2003, vol. 41.
- 34 G. Kresse and J. Hafner, Phys. Rev. B: Condens. Matter Mater. Phys., 1993, 47, 558.

35 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, 54, 11169.

36 P. E. Blöchl, Phys. Rev. B, 1994, 50, 17953.

Materials Advances

- 37 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, 77, 3865.
- 38 J. Heyd, G. E. Scuseria and M. Ernzerhof, *J. Chem. Phys.*, 2003, **118**, 8207–8215.
- 39 H. J. Monkhorst and J. D. Pack, *Phys. Rev. B: Solid State*, 1976, 13, 5188–5192.
- 40 R. King-Smith and D. Vanderbilt, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, 47, 1651.
- 41 R. Resta, Rev. Mod. Phys., 1994, 66, 899.
- 42 N. A. Spaldin, J. Solid State Chem., 2012, 195, 2-10.
- 43 G. F. Koster, J. O. Dimmock, R. G. Wheeler and H. Statz, *The Properties of the Thirty-Two Point Groups (Research Monograph)*, 1963.
- 44 H. T. Stokes and D. M. Hatch, *J. Appl. Crystallogr.*, 2005, 38, 237–238.
- 45 M. I. Aroyo, J. M. Perez-Mato, C. Capillas, E. Kroumova, S. Ivantchev, G. Madariaga, A. Kirov and H. Wondratschek, Z. Kristallogr. - Cryst. Mater., 2006, 221, 15–27.
- 46 Y. Hinuma, G. Pizzi, Y. Kumagai, F. Oba and I. Tanaka, *Comput. Mater. Sci.*, 2017, **128**, 140–184.
- 47 Wolfram Research, Inc., Mathematica, Version 12.3.1, https://www.wolfram.com/mathematica, Champaign, IL, 2021.
- 48 U. Herath, P. Tavadze, X. He, E. Bousquet, S. Singh, F. Munoz and A. H. Romero, *Comput. Phys. Commun.*, 2020, 251, 107080.
- 49 H. J. Zhao, P. Chen, C. Paillard, R. Arras, Y.-W. Fang, X. Li, J. Gosteau, Y. Yang and L. Bellaiche, *Phys. Rev. B*, 2020, 102, 041203.

- 50 H. J. Zhao, H. Nakamura, R. Arras, C. Paillard, P. Chen, J. Gosteau, X. Li, Y. Yang and L. Bellaiche, *Phys. Rev. Lett.*, 2020, 125, 216405.
- 51 C. Ederer and N. A. Spaldin, Phys. Rev. Lett., 2005, 95, 257601.
- 52 D. Ricinschi, K.-Y. Yun and M. Okuyama, *J. Phys.: Condens. Matter*, 2006, **18**, L97–L105.
- 53 Y. Heon Kim, A. Bhatnagar, E. Pippel, M. Alexe and D. Hesse, *J. Appl. Phys.*, 2014, **115**, 043526.
- 54 Bilbao Crystallographic Server, Generators and general positions, https://www.cryst.ehu.es/cryst/get_gen.html.
- 55 Bilbao Crystallographic Server, Point group tables, https://www.cryst.ehu.es/rep/point.html.
- 56 S. Vajna, E. Simon, A. Szilva, K. Palotas, B. Ujfalussy and L. Szunyogh, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, 85, 075404.
- 57 S. Bandyopadhyay and I. Dasgupta, *Phys. Rev. B*, 2021, 103, 014105.
- 58 J. M. Rondinelli and N. A. Spaldin, *Adv. Mater.*, 2011, 23, 3363–3381.
- 59 L. Tao and J. Wang, J. Appl. Phys., 2016, 120, 234101.
- 60 R. Arras, J. H. Gosteau, C. Paillard, Y. Yang and L. Bellaiche, Phys. Rev. B, 2019, 100, 174415.
- 61 H. J. Zhao, C. Xu, Y. Yang, W. Duan, X. M. Chen and L. Bellaiche, J. Phys.: Condens. Matter, 2015, 27, 485901.
- 62 H. J. Zhao, C. Xu, Y. Yang, W. Duan, X. M. Chen and L. Bellaiche, *J. Phys.: Condens. Matter*, 2015, 27, 485901.
- 63 C. Capillas, E. S. Tasci, G. de la Flor, D. Orobengoa, J. M. Perez-Mato and M. I. Aroyo, Z. Krist., 2011, 226(2), 186–196.
- 64 M. Ye and D. Vanderbilt, Phys. Rev. B, 2016, 93, 134303.