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## Investigation of the preparation and reactivity of metal–organic frameworks of cerium and pyridine-2,4,6-tricarboxylate†

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The synthesis of three coordination polymers of cerium(III) and the ligand pyridine-2,4,6-tricarboxylate (PTC) is reported. Two of the materials crystallise under hydrothermal conditions at 180 °C, with  $[\text{Ce}(\text{PTC})(\text{H}_2\text{O})_2] \cdot 1.5\text{H}_2\text{O}$ , (1), being formed on extended periods of reaction time, 3 days or longer, and  $\text{Ce}(\text{PTC})(\text{H}_2\text{O})_3$ , (2), crystallising after 1 day. Both phases contain Ce(III) but are prepared using the Ce(IV) salt  $\text{Ce}(\text{SO}_4)_2 \cdot 4\text{H}_2\text{O}$  as reagent. Under solvothermal conditions (mixed water-*N,N*-dimethylformamide (DMF)), the phase  $[\text{Ce}(\text{PTC})(\text{H}_2\text{O})(\text{DMF})] \cdot \text{H}_2\text{O}$  (3) is crystallised. The structures of the three materials are resolved by single crystal X-ray diffraction, with the phase purity of the samples determined by powder X-ray diffraction and thermogravimetric analysis. (1) is constructed from helical chains cross-linked by the PTC linkers to give a three-dimensional structure that contains clusters of water molecules in channels that are hydrogen-bonded to each other and to additional waters that are coordinated to cerium. (2) also contains nine-coordinate cerium but these are linked to give a dense framework, in which water is directly coordinated to cerium. (3) contains corner-shared nine-coordinate cerium centres, linked to give a framework in which Ce-coordinated DMF fills space. Upon heating the material (1) in air all water is irreversibly lost to give a poorly crystalline anhydrous phase Ce(PTC), as deduced from X-ray thermodiffractometry and thermogravimetric analysis. The material (1), however, is hydrothermally stable, and is also stable under oxidising conditions, where immersion in 30%  $\text{H}_2\text{O}_2$  gives no loss in crystallinity. Oxidation of around 50% of surface Ce to the +4 oxidation state is thus possible, as evidenced by X-ray photoelectron spectroscopy, which is accompanied by a colour change from yellow to orange. Photocatalytic activity of (1) is screened and the material shows effective degradation of methyl orange.

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## Introduction

Metal–organic frameworks of cerium have attracted significant interest in the past few years.<sup>1</sup> Part of this focus lies in the fact that cerium is one of the few lanthanide elements that adopts two stable oxidation states that can interconvert readily. The redox chemistry associated with Ce(IV)/Ce(III) finds uses

ranging from oxidants in organic synthesis,<sup>2</sup> to heterogeneous catalysis,<sup>3</sup> and integrating these properties into MOFs is an interesting proposition for development of new functional solids that might have practical applications. In the case of MOFs, a variety of Ce(III) materials have been produced and these are often analogues of structures that contain other larger lanthanides, such as those of La(III) or Nd(III). As expected, high coordination numbers are observed, often with irregular geometry and this can give unique structures, not found for metal cations from other parts of the Periodic Table. In the case of Ce(IV) the first MOFs formed were analogues of Zr(IV) materials, or mixed-metal variants where solid-solutions of Ce and Zr are found in an isostructural series.<sup>4</sup> It is noteworthy that the chemistry of Ce(IV) MOFs is largely associated with the hexameric oxy-clusters seen in zirconium chemistry, whereas for Ce(III), a variety of coordination geometries and connectivity of the primary polyhedra is seen, depending on the ligands used to construct the material.<sup>1a</sup>

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A variety of carboxylate linkers have been used for the preparation of cerium MOFs, each with inherent structural directing connectivity, depending on the number of connection points, and also with the possibility of structural flexibility, depending on the organic backbone of the ligand.<sup>1a</sup> Carboxylates provide versatile ligands for construction of MOFs based on lanthanide cations because the variety of potential bonding modes of the carboxylate function allows the high coordination number of the cation to be satisfied,<sup>5</sup> as well as the strong electrostatic contribution to bonding leading to thermally, and potentially hydrothermally, robust materials.

In terms of properties, the MOFs reported for cerium have been studied for a range of potential applications. The reported works span catalysis, photocatalysis and sensing, as well as gas adsorption that would be associated with many other families of MOFs. For example, the cerium(iv) analogue of UiO-66 was used as an oxidation catalyst, proven with the model reaction of conversion of benzyl alcohol to benzaldehyde, and good conversion was reported even though the reaction was found to occur only at the surface of the material.<sup>6</sup> Homochiral Ce(III) MOFs produced using methyl-enediisophthalate as linker were found to give enantiomeric selectivity in the asymmetric cyanosilylation of carbonyl-containing substrates, and were easily recovered by filtration and reused without loss of efficiency.<sup>7</sup> Application in photocatalysis were predicted by computational simulation for Ce(iv)-UiO-66, which showed that only the Ce analogue of material could allow ligand-to-metal charge transfer to separate photogenerated charges.<sup>8</sup> Photo-oxidation of benzylic alcohols to benzaldehydes under near-ultraviolet light irradiation was independently proven for the material.<sup>9</sup> In sensing applications a chromogen has been used in combination with Ce-MOFs to detect species such as ascorbic acid, using a mixed-valent Ce(III)/Ce(IV) MOF,<sup>10</sup> and Hg<sup>2+</sup> using Ce(iv)-UiO-66.<sup>11</sup> The emerging applications of Ce-MOFs have recently been reviewed.<sup>1</sup>

In this paper we describe an exploration of the synthesis of metal-organic frameworks of cerium using the linker pyridine-2,4,6-tricarboxylate (PTC). This linker has been used for a number of metal organic frameworks,<sup>12</sup> but is worthy of further investigation since tridentate binding by the nitrogen and two neighbouring carboxylate oxygens can potentially provide robust materials, with the third carboxylate moiety offering the possibility of bridging to further metal centres to create three-dimensionally extended structures. Das *et al.* reported eleven possible bonding bonds for the PTC ligand.<sup>12</sup> Here, we report three distinct Ce(III)-PTC materials and for one of the materials, prepared in water with extended crystallisation times, we study its stability and explore possible applications. A structural comparison is made with previously reported lanthanide pyridine-2,4,6-tricarboxylate materials.

## Experimental section

The synthesis of the ligand precursor pyridine-2,4,6-tricarboxylic acid, H<sub>3</sub>PTC, was carried out by oxidation of 2,4,6-tri-

methylpyridine and recrystallisation *via* formation of the methyl ester of the tricarboxylic acid. 2,4,6-Trimethylpyridine (6 mL, 43 mmol) was added to concentrated H<sub>2</sub>SO<sub>4</sub> (200 mL) at 0 °C followed by small amounts of CrO<sub>3</sub> (39 g, 0.39 mmol, in total) with the temperature maintained at 0 °C. The mixture was heated to 75 °C for 24 h, then placed into ice water (0.5 mL) and kept at 4 °C. After 3 days the mixture was filtered and the filtrate was placed in 40 mL of methanol and 0.7 mL of H<sub>2</sub>SO<sub>4</sub>. After 12 h of reflux, the solid methyl ester of pyridine-2,4,6-tricarboxylic acid was recovered by filtration with a yield of 60%. Hydrolysis of the ester was performed in 30 mL of water and 30 mL of 38% hydrochloric acid, and the solution was heated until complete dissolution (1 h). At room temperature the pH of the solution was adjusted to 2 and the precipitated H<sub>3</sub>PTC crystals were filtered and washed with ice-cold H<sub>2</sub>O (50 mL) to give a yield of 80%.

The cerium precursors Ce(SO<sub>4</sub>)<sub>2</sub>·4H<sub>2</sub>O and Ce(NO<sub>3</sub>)<sub>3</sub>·4H<sub>2</sub>O were used as provided by chemical suppliers, with the water content verified using thermogravimetric analysis. [Ce(PTC)(H<sub>2</sub>O)<sub>2</sub>]·1.5H<sub>2</sub>O (1) was isolated as single crystals suitable for structure determination in a hydrothermal reaction. Ce(SO<sub>4</sub>)<sub>2</sub>·4H<sub>2</sub>O (0.101 g, 0.25 mmol) and H<sub>3</sub>PTC (0.106 g, 0.5 mmol) were dispersed in 5 mL H<sub>2</sub>O with rapid stirring and 4 drops (~0.2 mL) of triethylamine were added to aid deprotonation of the tricarboxylic acid. The mixture was sealed in a 23 mL PTFE-lined steel autoclaved and heated in a fan-assisted oven 180 °C for 10 days. The solid product was recovered by suction filtration and allowed to dry under ambient conditions. Ce(PTC)(H<sub>2</sub>O)<sub>3</sub> (2) was formed using identical conditions to (1), except the time of reaction was shortened to 1 day. [Ce(PTC)(H<sub>2</sub>O)(DMF)]·H<sub>2</sub>O (3) was prepared at room temperature in a mixed solvent system. 0.0211 g of H<sub>3</sub>PTC was dissolved in 5 mL of 1 M aqueous NaOH and the 5 mL DMF (*N,N*-dimethylformamide) was added. Separately, 5 mL 0.02 M aqueous Ce(NO<sub>3</sub>)<sub>3</sub>·4H<sub>2</sub>O was diluted with a further 5 mL water. The second solution was added dropwise to the first one and the mixture allowed to stand. After 30 days crystals suitable for single-crystal diffraction were isolated.

Single crystal X-ray diffraction was performed on suitable crystals that were selected and mounted on glass fibres with Fromblin oil and placed on a Rigaku Oxford Diffraction SuperNova diffractometer with a dual source (Cu at zero) equipped with an AtlasS2 CCD area detector. The crystals were kept at 150(2) K during data collection. Using Olex2,<sup>13</sup> the structures were solved with the ShelXS structure solution program<sup>14</sup> using the Patterson Method and refined with the ShelXL refinement package<sup>15</sup> using least squares minimisation. The ESI† contains further details of the structure solution and crystal structure data.

Powder XRD data for phase identification were recorded using a Siemens D5000 X-ray diffractometer operating with Cu K $\alpha$ 1/2 radiation. Data were recorded in Bragg–Brentano geometry from finely-ground samples pressed into silicon holders.

Thermogravimetric analysis was performed using a Mettler-Toledo TGA/DSC 1 Thermogravimetric Analyser. Approximately 10 mg of powdered sample was placed in an alumina pan and



its mass monitored in static air upon heating to 1000 °C at a rate of 10 °C min<sup>-1</sup>.

Powder X-ray thermodiffractometry was performed using a Bruker D8 instrument with Cu K $\alpha$ 1/2 radiation, fitted with an Anton Paar XRD 900 chamber and a VÄNTAC solid-state detector. Samples were heated from room temperature to 900 °C in flowing air and XRD patterns were recorded at intervals of 50 °C.

Diffuse reflectance UV-Vis spectroscopy was performed on powder samples using a Shimadzu UV-2600i UV-Vis spectrophotometer. A barium sulfate standard was used as a baseline for the measurements.

X-ray photoelectron spectroscopy (XPS) was performed on a Kratos Axis Ultra DLD using monochromatic Al-K $\alpha$  excitation (1486.7 eV) with detection at 90° with respect to the surface. A charge neutraliser was used during the experiments in order to prevent surface charging, with the binding energy scale subsequently referenced to the C-C/C-H component of the C 1s spectrum at 285.0 eV during data analysis. Data were analysed using the CasaXPS package applying Voigt peaks and Shirley backgrounds for all regions. The Ce 3d region was interpreted using previous literature reports on cerium oxides as guidance.<sup>16</sup>

Photocatalytic testing was carried out using a Hg vapour lamp (250 W) with 2 mg solid catalyst and 10 mL of a 4.0 × 10<sup>-5</sup> M solution of methyl orange at pH = 3.0. The concentration of dye was monitored at 505 nm using UV-Vis spectroscopy as a function of time on a Hewlett Packard HP 8453 spectrophotometer.

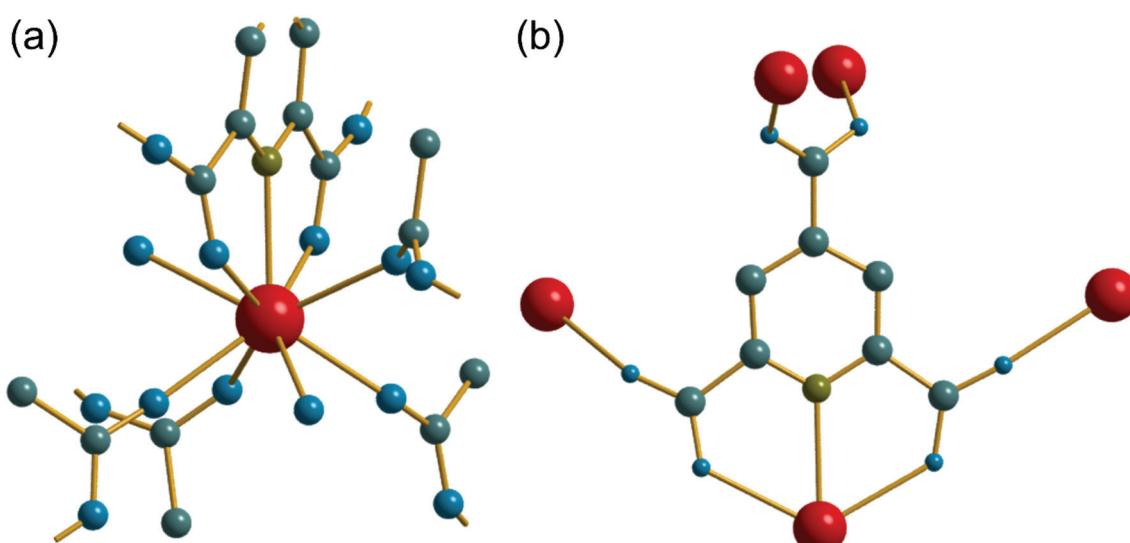
## Results and discussion

The synthesis of two of the cerium organic frameworks (1) and (2) was carried out under hydrothermal conditions, with the only difference being the extended reaction time to form (1).

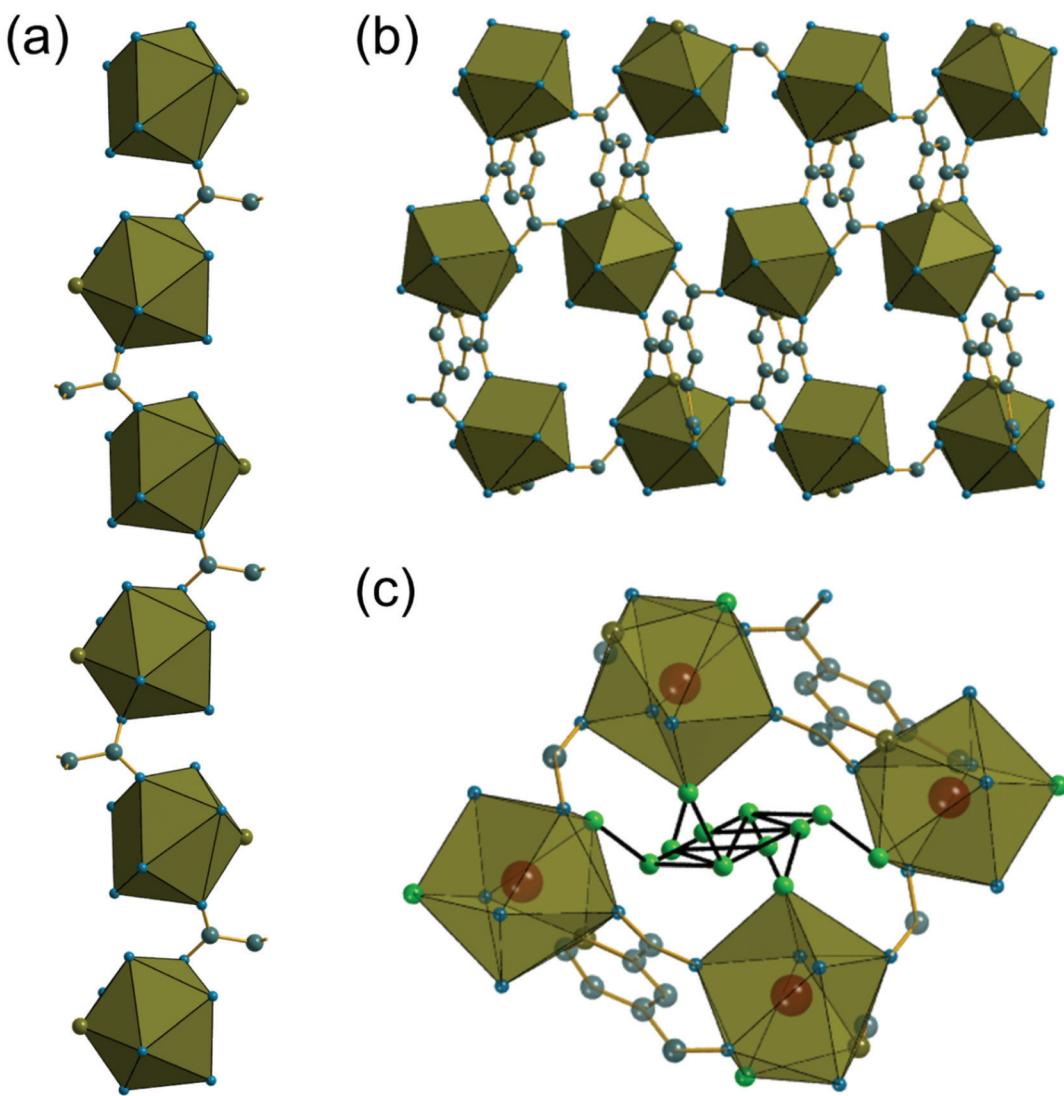
The metal-organic framework structure of (1) is the same as reported for Nd<sup>3+</sup> and Pr<sup>3+</sup> materials,<sup>17</sup> and already found for Ce<sup>3+</sup>,<sup>18</sup> but (1) contains a different amount of occluded water than the previously reported structures. The previously reported structure of the cerium material by Zhao *et al.* has chemical formula [Ce(PTC)(H<sub>2</sub>O)<sub>2</sub>]·H<sub>2</sub>O.<sup>18</sup> For the Nd analogue a chemical formula of [Nd(PTC)(H<sub>2</sub>O)<sub>2</sub>]·2H<sub>2</sub>O was reported by Ghosh and Bharadwaj,<sup>17a</sup> whereas for Pr, isostructural [Pr(PTC)(H<sub>2</sub>O)<sub>2</sub>]·2H<sub>2</sub>O<sup>16a</sup> and [Pr(PTC)(H<sub>2</sub>O)<sub>2</sub>]·1.5H<sub>2</sub>O<sup>17b</sup> materials have been reported (Table S2†). This leads us to conclude that variable water content is possible. The structure of (1) is constructed from 9-coordinate Ce centres created from coordination of the PTC ligand and water molecules, Fig. 1a. Three of these coordinating atoms are from tridentate coordination of the PTC ligand *via* nitrogen and two oxygens from neighbouring carboxylates, four are oxygens of other carboxylate, and two are oxygens of water molecules. Each PTC ligand thus connects five Ce centres, Fig. 1b. This bonding mode PTC is Type VII according to the classification by Das *et al.* (ESI†).<sup>12</sup>

The 9-coordinate Ce centres are not directly corner shared in (1) but are most closely linked *via* Z,Z- $\mu_2$ - $\eta^1$ .. $\eta^1$  bridging carboxylate oxygens to form helical chains that run parallel to *b*, Fig. 2a. There are chains of each handedness in the crystal structure. The chains are cross-linked in both directions by to give a three-dimensionally connected framework, Fig. 2b. Running parallel to *a* are channels that contain occluded water, as well as that directly coordinated to cerium. The pores present can be defined by closest distances between the water oxygens directly bound to cerium of 3.29 Å, and in this confined space the occluded water is held by a network of hydrogen bonding, as evidenced by short O–O distances (Fig. 2c and Table S7†), which includes interactions between the occluded water and the water that is coordinated to the cerium directly.

The material (2) contains two crystallographically distinct Ce sites, both of which have coordination number nine and



**Fig. 1** (a) The local environment of Ce in (1) with cut-off bonds showing connection to neighbouring atoms. (b) The connectivity of the PTC ligand in (1). In both cases, red spheres are cerium, grey are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown.



**Fig. 2** (a) Chains of nine-coordinate Ce centres found in (1) running parallel to *b*. (b) Projection of the structure in the *ab* plane showing connection of chains to yield an open structure (extra-framework water molecules not shown). (c) The local environment of water in (1) with short oxygen–oxygen distances (2.5 to 3 Å) indicative of hydrogen bonding shown in black. Note that the extra framework water sites are not fully occupied so not all will be simultaneously populated. The cerium environment is represented by the polyhedra, and grey spheres are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown. In (c) the green spheres are the oxygens of water molecules.

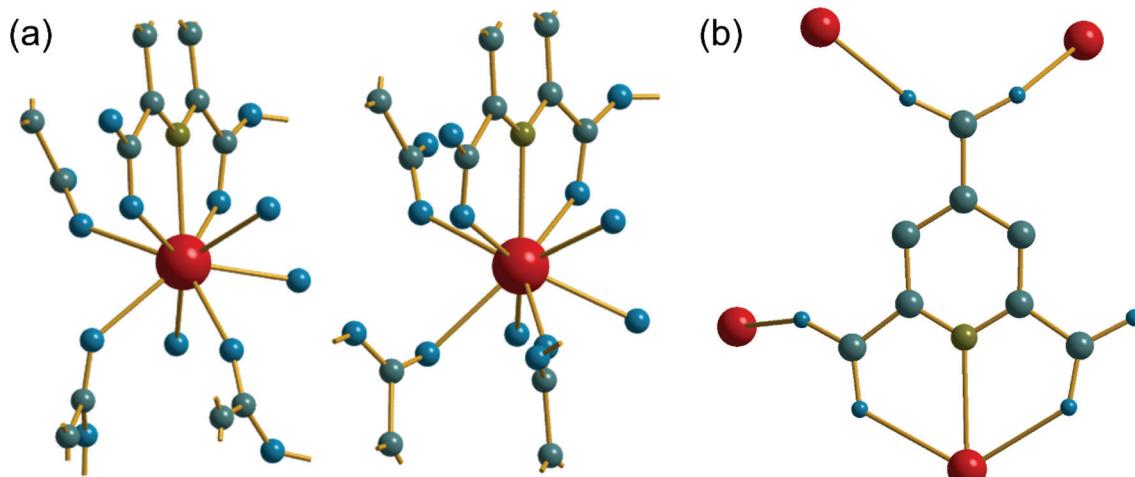
the same set of coordinating atoms, Fig. 3a. These consist of one tridentate PTC ligand bound through a nitrogen and two neighbouring carboxylate oxygens, three further carboxylate oxygens from distinct PTC ligands and three water molecules. Each PTC ligand, of two crystallographically distinct types, connects four Ce centres in (2), as shown in Fig. 3b, such that one of the carboxylate oxygens is terminal: this is a mode of bonding not described in the Das *et al.* notation.<sup>12</sup>

As in (1), the 9-coordinate Ce(III) centres in (2) are not connected by directly shared corners, and here are linked to the neighbouring cerium centres *via*  $Z,Z\text{-}\mu_2\text{-}\eta^1\text{:}\eta^1$  bridging carboxylates to give a three-dimensionally connected array. Each Ce centre is thus connected to four other centres and this occurs in a strictly alternating manner with one of the two crystallo-

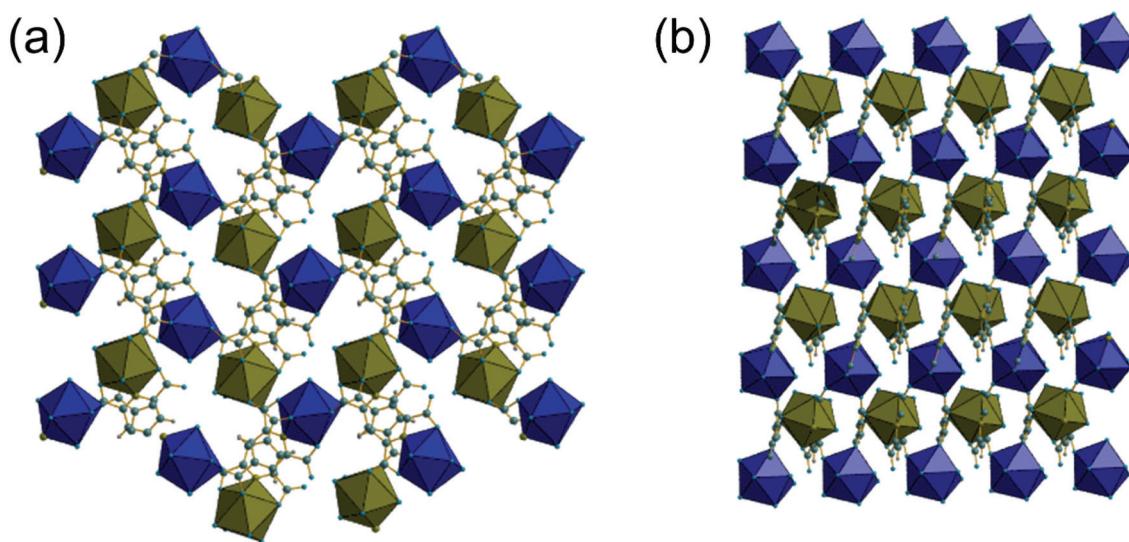
graphically distinct cerium centres surrounded only by the second type and *vice versa*, as seen when the structure is viewed along *b*, Fig. 4a. The structure is more densely packed than (1) but the terminal carboxylate oxygens project into small voids, seen when the structure is viewed along *a*, Fig. 4b. These oxygens lie within 3 Å of oxygens of water molecules connected directly to more distant cerium centres, meaning that there is no possibility of permanent porosity in (2), indeed there is no occluded water that might be removed to access any free pore space.

The phase purity of (1) and (2) was determined by powder XRD, Fig. 5a. This showed that the bulk sample of (2) is contaminated with just a small quantity of (1). Powder XRD patterns of materials prepared at intermediate reaction times are





**Fig. 3** (a) The local environment of the two Ce sites in (2) with cut-off bonds showing connection to neighbouring atoms. (b) The connectivity of one of the PTC ligands in (2); the second PTC ligand has identical connectivity and is not shown. In both cases, red spheres are cerium, grey are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown.



**Fig. 4** (a) View of (2) along *b* with the two distinct types of cerium shown in different coloured polyhedra (Ce1 blue and Ce2 olive green). (b) View of (2) along *a* showing the dense packing of the structure. Grey spheres are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown.

also shown on Fig. 5, which reveals an increasing fraction of (1) over (2) with increasing time of synthesis. At 72 hours the sample is phase pure (1) (note that the crystal studied for structure solution was grown using an even more extended reaction time of 10 days).

Given that the two structures bear no obvious relationship in atomic connectivity, it is likely that the successive crystallisation of (2) followed by (1) occurs *via* dissolution of the initially formed (2). In other MOF chemistry, the formation of kinetic phases has been observed, that may have no structural relationship to the final crystalline product.<sup>19</sup> The presence of Ce<sup>3+</sup> in (1) and (2) is assigned to achieve charge balance, but

also verified using bond valence sums,<sup>20</sup> which give an oxidation state of 3.23 for the single unique cerium in (1) and 3.14 and 3.27 for Ce1 and Ce2, respectively, in (2). The reduction of the Ce<sup>4+</sup> precursor used for both materials must take place readily in solution. It has previously been noted that Ce<sup>4+</sup> MOFs are only produced under solvothermal reaction conditions from (NH<sub>4</sub>)<sub>2</sub>[Ce(NO<sub>3</sub>)<sub>6</sub>] in water/DMF mixtures at low reaction temperatures  $\leq 100$  °C using short reaction times ( $t \leq 30$  min).<sup>1a</sup> In other cases where Ce(IV) precursors have been used, reduction to Ce(III) occurs readily *in situ* to yield materials that contain solely the lower oxidation state. For example, in the case of porphyrin-based MOFs, the use of



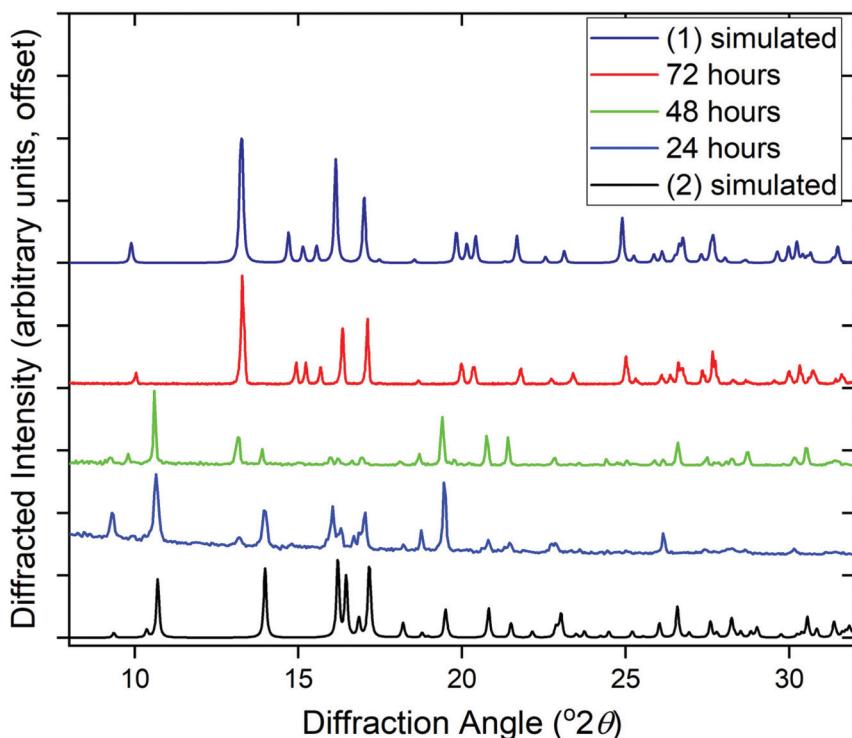


Fig. 5 Powder XRD patterns of cerium pyridine-2,4,6-tricarboxylates prepared at different reaction times compared to simulated patterns of (1) and (2), showing the evolution of the product formed from (2) to (1).

$(\text{NH}_4)_2[\text{Ce}(\text{NO}_3)_6]$  lead to Ce(III) materials, which was ascribed to the reducing properties of the ligand precursor.<sup>21</sup> We also attempted reactions using NaOH as base but this yielded only crystals of the sodium salt  $\text{Na}_3\text{PTC}$ , as verified by single crystal diffraction (see ESI†).

The material (3) is formed at room temperature from a mixed DMF-water solvent using a  $\text{Ce}^{3+}$  containing precursor. The crystal structure contains one unique Ce centre, which is directly coordinated to nine ligand atoms, Fig. 6a. Three of these are from the tridentate coordination of a PTC ligand *via* its nitrogen and neighbouring oxygens of adjacent carboxylate groups. A further PTC ligand is connected *via* two oxygens in an  $\eta^2$  binding mode, while a third PTC ligand is connected by a single oxygen atom. Two directly coordinated water molecules and an O-connected DMF molecule complete the coordination environment. The single unique PTC ligand thus connects a total of four different Ce centres, as shown on Fig. 6b, such that one of the carboxylate oxygens is terminal, as in (2). This mode of binding is not one reported in the Das *et al.* classification.<sup>12</sup>

Unlike compounds (1) and (2), the cerium centres in (3) directly share a common oxygen *via* a  $\mu_2\text{-}\eta^2\text{-}\eta^1$  connected PTC ligand, using one of the carboxylates adjacent to the nitrogen of the pyridine ring. This gives infinite chains running parallel to *c*, Fig. 7a. The carboxylate in the 4-position of the pyridine ring then cross links the chains by connection to two ceriums on a neighbouring chain in a *Z,Z*- $\mu_2\text{-}\eta^1\text{-}\eta^1$  fashion. The  $\text{Ce}-\text{O}-\text{Ce}$  bond angle along the chain is  $138^\circ$  and so the chains are

not linear: there is a zig-zag arrangement of cerium centres that repeats every second cerium, evident in a projection parallel to *c*, Fig. 7b, which shows the connection of the chains to each other. The DMF molecules, coordinated to the Ce centres, fill any void space that may otherwise be present, and are found to be disordered, modelled as being present in two different orientations, with occupancy of 70 : 30.

Two other cerium(III) pyridine-2,4,6-tricarboxylates have already been reported in the literature.  $[\text{Ce}(\text{PTC})(\text{H}_2\text{O})_3]\cdot\text{H}_2\text{O}$  was reported by Sharif *et al.* and contains a single unique Ce coordinated by 9 ligand atoms, but has a different local coordination environment to our materials, since two of the ligand positions around Ce are formed from  $\eta^2$  binding mode from a single PTC ligand, which is coordinated in Type XI mode using the Das *et al.* notation. This material was prepared in a hydrothermal reaction in the presence of trifluoroacetic acid. The material  $[\text{Ce}(\text{PTC})(\text{H}_2\text{O})_5]\cdot 4\text{H}_2\text{O}$ , also reported by Sharif *et al.*, is much more water-rich and contains five directly coordinated water molecules to each nine-coordinate Ce centre.<sup>22</sup> The PTC ligand is bound in a manner not classified by Das *et al.* where two cerium centres are connected by each PTC ligand (see ESI†). This material was produced in water-methanol mixture at room temperature. Other lanthanide pyridine-2,4,6-tricarboxylates are known, and for the later lanthanides the smaller size of the  $\text{Ln}^{3+}$  cations with increasing atomic number may result in lower coordination numbers, even for an isostructural series, where the water content may be variable.<sup>23</sup> We also note that PTC has been used as a ligand in combi-



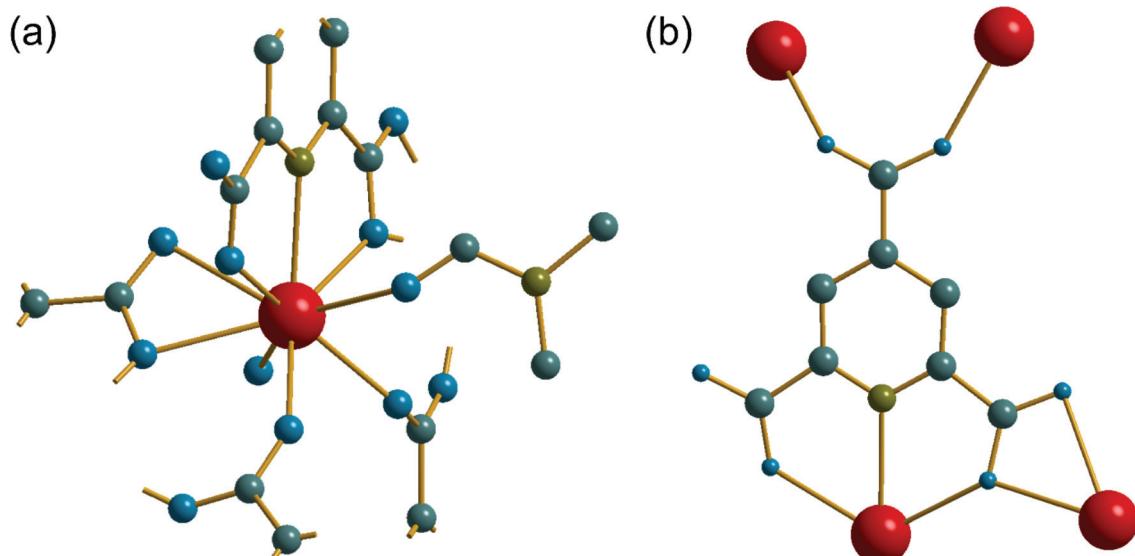


Fig. 6 (a) The local environment of Ce in (3) with cut-off bonds showing connection to neighbouring atoms. (b) The connectivity of the PTC ligand in (3). In both cases, red spheres are cerium, grey are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown.

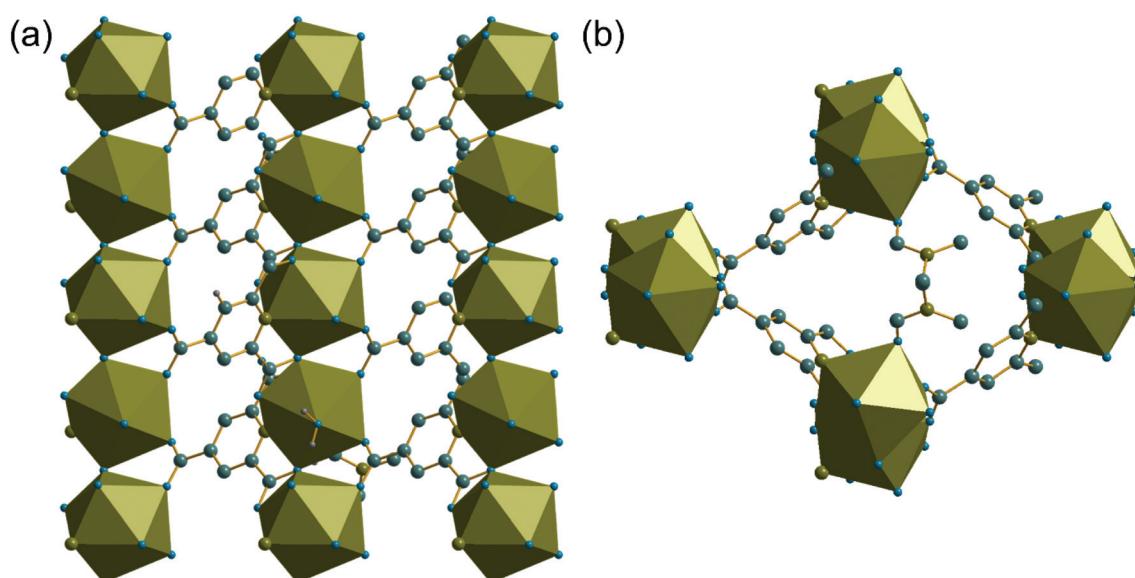


Fig. 7 (a) View of (3) along *b* showing cross-linked chains of corner-shared cerium-centred polyhedra. (b) Projection of (3) along *c* showing co-ordinated DMF (one orientation shown) filling extra-framework space. The cerium environment is represented by the polyhedra, and grey spheres are carbon, blue are oxygen, olive are nitrogen and hydrogens are not shown.

nation with a second ligand to form extended coordination networks with cerium, such as with oxalate or glycine,<sup>24</sup> but structural comparison with these materials is not straightforward due to the influence of the second ligand on the architecture of the structure.

For each of the materials studied here, thermogravimetric analysis, in addition to powder XRD, confirmed phase purity and showed stepwise loss of occluded water, coordinated water (or DMF where present), before final combustion of the PTC

ligand (ESI†). Since (1) is the material produced under hydrothermal conditions for extended periods of heating, it represents the most stable phase in the this set of materials and we further studied its properties. Thermodiffractometry shows the material maintains crystallinity upon heating to 100 °C, with no evident shift in Bragg peak positions or broadening of Bragg peaks, Fig. 8. Above 110 °C a significant change in the powder XRD pattern is seen with evidence of a phase transition, but also a broadening of all Bragg peaks indicating a

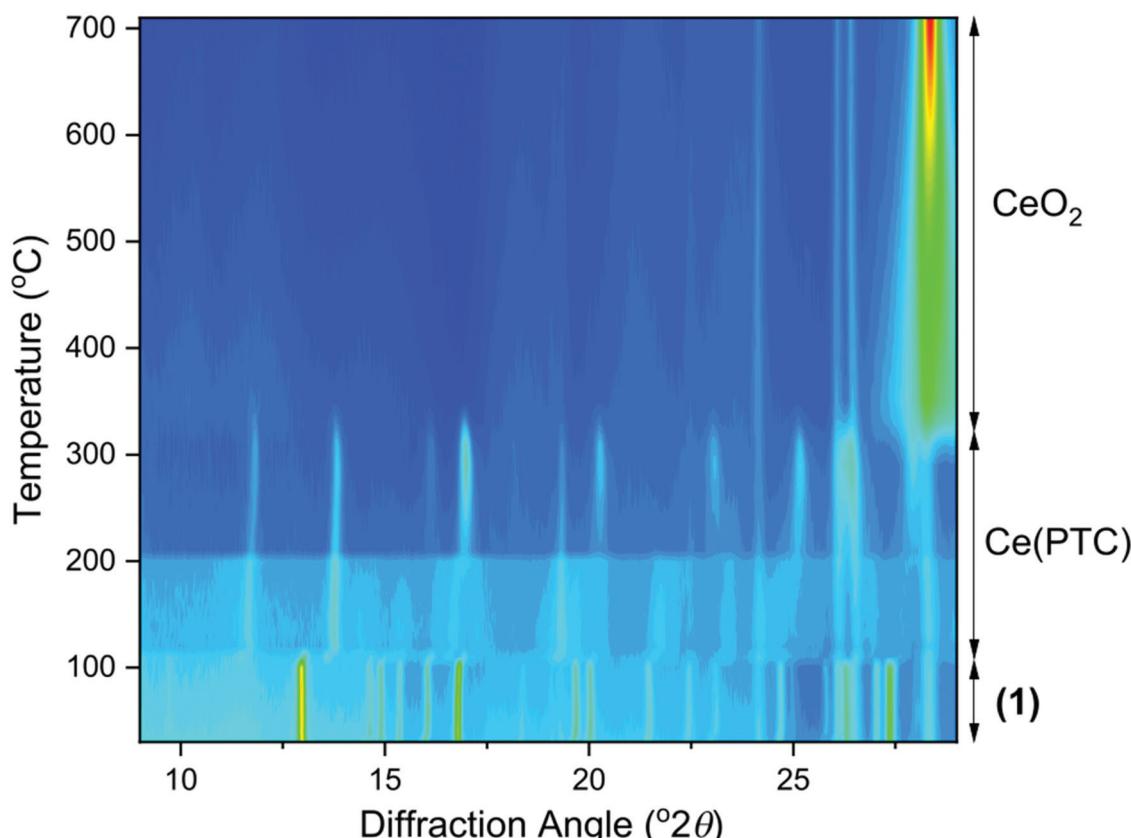


Fig. 8 Thermodiffractometry measured for (1) on heating in air, with the phases present indicated.

loss of crystallinity. The phase produced is stable to 310 °C. Looking at the TGA analysis, the phase formed between 110 °C and 310 °C would correspond to fully dehydrated Ce(PTC), where loss of occluded water and directly bound water has occurred. The TGA shows no evidence that the two types of water can be removed separately, implying that the extra-framework water present plays a significant role in stabilising the open structure through its hydrogen bonds. It is noteworthy that the poorly crystalline Ce(PTC) phase found between 110 and 310 °C, shows no low-angle diffraction features (ESI†), implying that the structure is more dense than (1). Its BET surface area is also small,  $\sim 7 \text{ m}^2 \text{ g}^{-1}$ , consistent with the formation of a dense structure. In addition, the anhydrous phase does not convert back to (1) on standing in ambient air or immersion in water, consistent with significant structural rearrangement having taken place upon dehydration. Attempts to remove the water by vacuum treatment were also unsuccessful and the same collapsed Ce(PTC) phase was observed by powder XRD (ESI†).

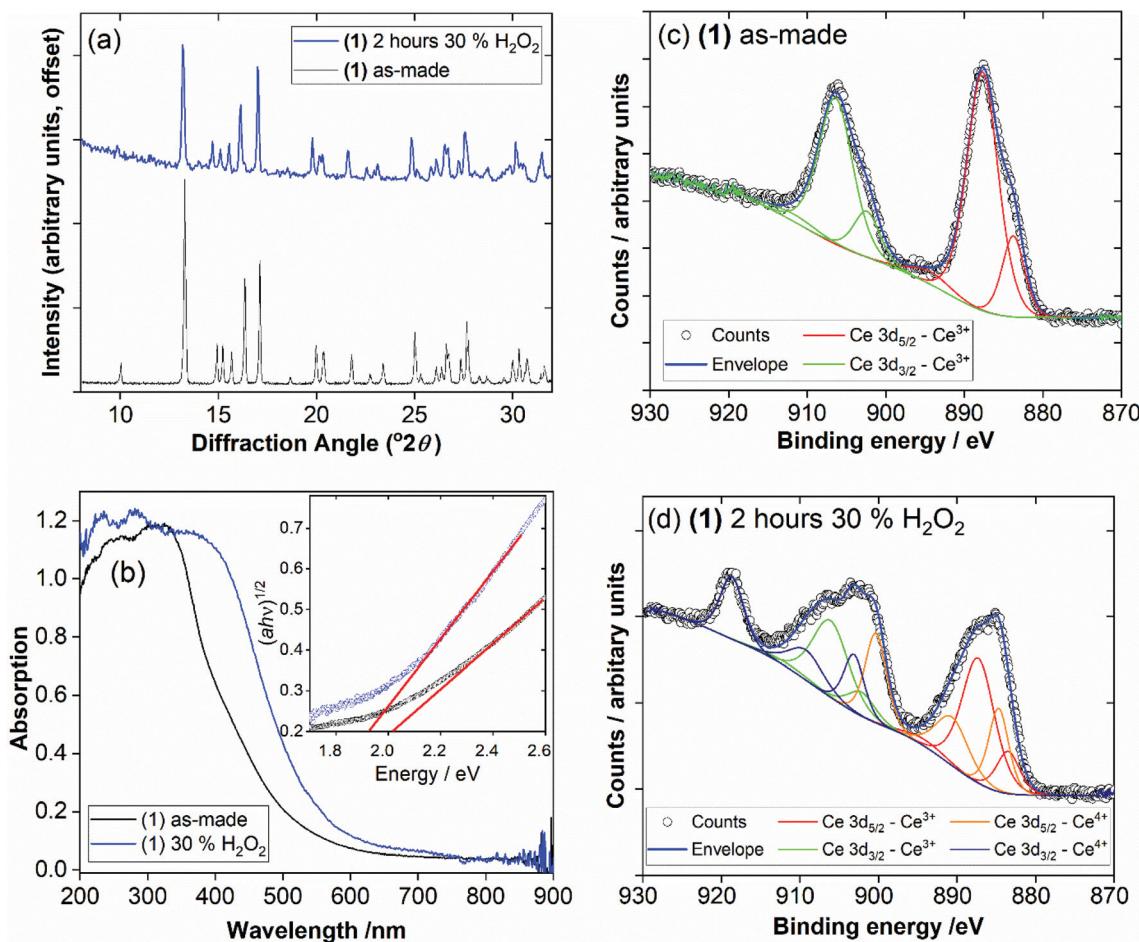
Upon heating above 310 °C, further structural collapse is evident, with the first appearance of the (111) Bragg peak of CeO<sub>2</sub> that subsequently increases in intensity and sharpens on continued heating. Thus above 310 °C, combustion of the organic ligand commences, which corresponds with the TGA analysis.

The possibility of Ce<sup>3+</sup>/Ce<sup>4+</sup> redox is useful to explore in coordination polymers. We thus assessed the stability and

behaviour of (1) towards hydrogen peroxide. After immersion of (1) in 30 v/v H<sub>2</sub>O<sub>2</sub> solution for 2 hours, the powder XRD pattern shows the same characteristic signature, with no changes in Bragg peaks positions and only small changes in relative peak intensities, Fig. 9a. The colour of the sample, however, is notably different, with a change from pale yellow to orange. This colour change was quantified using diffuse-reflection UV-Vis spectroscopy, which indicates a small decrease in band gap from  $\sim 2.0$  to  $\sim 1.9$  eV, Fig. 9b. XPS was used to examine the surface oxidation state of the material before and after oxidation, Fig. 9c and d. Quantification of the signal shows that around 50% of the Ce<sup>3+</sup> present at the surface is oxidised to Ce<sup>4+</sup> after oxidation. Given that the powder XRD shows little change and that XPS has a probe depth of a few nanometres, it is probable that the reduction of cerium is associated with the surface of the sample, and in turn responsible for the bulk colour change observed. TGA also shows no significant differences after the oxidation treatment, which again implies only the surface is modified (see ESI†). Xiong *et al.* previously studied a cerium(m) benzenetricarboxylate material that could be partially oxidised by treatment with H<sub>2</sub>O<sub>2</sub>,<sup>25</sup> and thus (1) behaves in a similar manner.

One interesting application of cerium-organic frameworks is in photocatalysis, and we assessed the possibility of (1) for this purpose, given its stability in aqueous conditions. The degradation of methyl orange was studied using a 250 W





**Fig. 9** Characterisation of (1) upon application of oxidising conditions: (a) powder XRD measured before and after exposure to 30%  $\text{H}_2\text{O}_2$  solution, (b) UV-Vis of the two solids, with Tauc plots inset, where red lines showing the linear extrapolation to estimate band gap, and (c) and (d) XPS of the materials, with the deconvolution and assignment of signals indicated.

mercury lamp as UV source. This showed an effective decomposition of the dye in the presence of (1) alone, which was enhanced with added  $\text{H}_2\text{O}_2$ , Fig. 10a. We also verified the decomposition of dye in the presence of (1) that had been previously oxidised. Adsorption effects were found to be negligible (below 0.2%), as shown by a background measurement in the absence of light (ESI†). A decrease in the intensity of the absorption band ( $\sim 505$  nm) relative to  $-\text{N}=\text{N}-$  double bond in the azo dye and the decolourisation of the solution indicate an efficient photodegradation (see ESI†). Also, for photocatalysis in the presence of our catalyst a new band around 280 nm appears and becomes stronger over reaction time, indicating the cleavage of the azo bond and the formation of structures with  $-\text{NH}_2$  groups.<sup>26</sup>

After 60 min, the degradation rates were 80%, 87% and 94% for (1) alone, previously oxidised (1) and (1) +  $\text{H}_2\text{O}_2$ . Degradation rates of methyl orange were estimated by dye decolourisation following eqn (1):

$$\text{Degradation}(\%) = \left[ \frac{C_0 - C}{C_0} \right] \times 100 \quad (1)$$

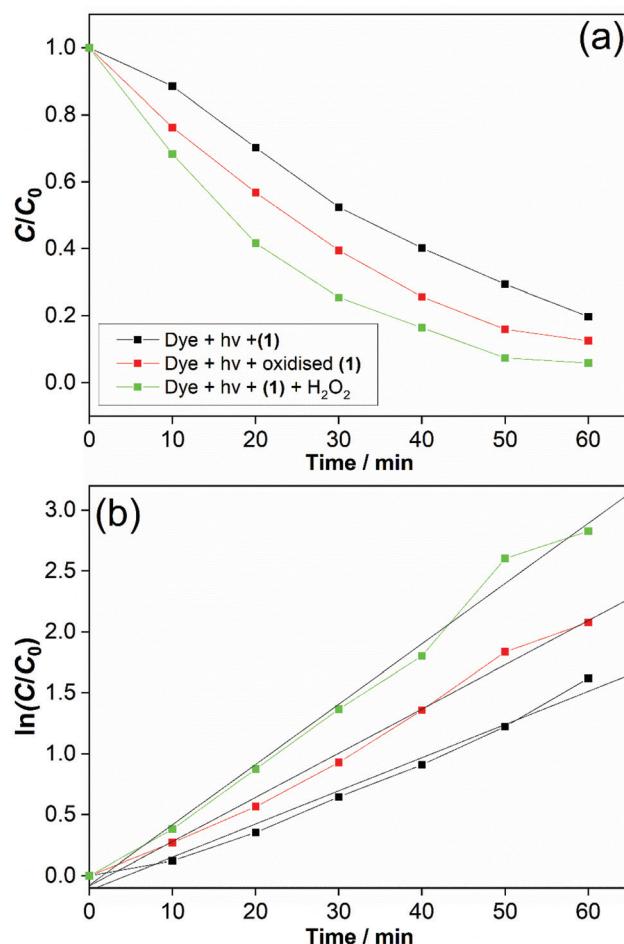
The photodegradation follows pseudo first-order kinetics, Fig. 10b, as expected for heterogeneous photocatalysis.<sup>27</sup> The

corresponding rate constants  $k_{\text{obs}}$  ( $\text{min}^{-1}$ ) for methyl orange were determined by pseudo first-order kinetic expression (eqn (2)). Fitted linear equation ( $R^2 = 0.982\text{--}0.992$ ) for all photocatalytic experimental conditions is showed in Fig. 10b.

$$\ln\left(\frac{C_0}{C}\right) = k_{\text{obs}}t \quad (2)$$

$k_{\text{obs}}$  were  $0.0272 \text{ min}^{-1}$ ,  $0.0363 \text{ min}^{-1}$  and  $0.0508 \text{ min}^{-1}$  for (1) alone, (1) previously oxidised and (1) +  $\text{H}_2\text{O}_2$ .

The photocatalytic performance of cerium-PTC based MOFs was significantly affected by the presence of surface  $\text{Ce}^{3+}/\text{Ce}^{4+}$ . Simultaneous presence of these species is known to favour redox reactions in the photocatalytic mechanisms.<sup>28</sup> The photodegradation of methyl orange was conducted in the presence of catalyst (1) and  $\text{H}_2\text{O}_2$  (0.01 M), simulating a Fenton-like process.<sup>29</sup> The photocatalysis was improved and best conversion was achieved with methyl orange almost completely degraded after 60 min. For this photodegradation the effect of hydrogen peroxide on the dye must be considered (ESI†). Although the photocatalysis was enhanced with  $\text{H}_2\text{O}_2$  addition, the performance of (1) alone and previously oxidised (1) must be highlighted. Very low quantities of catalysts were employed



**Fig. 10** (a) Photodegradation of methyl orange at catalysts, 25 °C, UV light (250 W). (b) Kinetic fit of the time trace and linear plots (pseudo first-order) of the UV-VIS band at  $\lambda_{\text{max}} = 505$  nm,  $R^2 = 0.982$  (black line), 0.992 (red line) and 0.987 (green line). The legend in (b) is the same as in (a).

in our experiments and good conversion rates were obtained. The reported works for dye degradation in the presence of metal-organic frameworks have typically described the use more than three-fold excess of catalyst per dye concentration.<sup>27b,30</sup> While CeO<sub>2</sub> has been employed as catalyst with similar results for photodegradation,<sup>28,31</sup> it is noteworthy that water-stable and oxidatively-stable cerium organic frameworks provide effective photocatalysts.

## Conclusions

We have explored the hydrothermal chemistry of cerium salts in combination with a polydentate ligand that includes a strongly binding tridentate binding motif, *via* a pyridine nitrogen and two adjacent carboxylate motifs. Despite using cerium (IV) precursors, the crystalline product are Ce(III) frameworks. It can be noted that the previous synthesis of the material isostructural to (1) used a Ce(III) precursor, illustrating the strong

tendency of cerium to form Ce(III) carboxylates. The variety of bonding modes of pyridine-2,4,6-tricarboxylate is illustrated by the materials we have crystallised, which add to a wider family of lanthanide pyridine-2,4,6-tricarboxylates already reported in the literature. The material (1) is an example of a hydrothermally-stable metal-organic framework, which is also stable under oxidative conditions, where evidence is found for oxidation of a large proportion of cerium at the surface to the +4 oxidation state.

## Conflicts of interest

There are no conflicts to declare.

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## References

- (a) J. Jacobsen, A. Ienco, R. D'Amato, F. Costantino and N. Stock, *Dalton Trans.*, 2020, **49**, 16551–16586; (b) Z. Hu, Y. Wang and D. Zhao, *Chem. Soc. Rev.*, 2021, **50**, 4629–4683.
- V. Sridharan and J. C. Menéndez, *Chem. Rev.*, 2010, **110**, 3805–3849.
- T. Montini, M. Melchionna, M. Monai and P. Fornasiero, *Chem. Rev.*, 2016, **116**, 5987–6041.
- K. A. Lomachenko, J. Jacobsen, A. L. Bugaev, C. Atzori, F. Bonino, S. Bordiga, N. Stock and C. Lamberti, *J. Am. Chem. Soc.*, 2018, **140**, 17379–17383.
- S. R. Batten, S. Neville and D. R. Turner, *Coordination Polymers: Design, Analysis and Application*, Royal Society of Chemistry, Cambridge, UK, 2009.
- M. Lammert, M. T. Wharmby, S. Smolders, B. Bueken, A. Lieb, K. A. Lomachenko, D. D. Vos and N. Stock, *Chem. Commun.*, 2015, **51**, 12578–12581.
- D. Dang, P. Wu, C. He, Z. Xie and C. Duan, *J. Am. Chem. Soc.*, 2010, **132**, 14321–14323.
- X.-P. Wu, L. Gagliardi and D. G. Truhlar, *J. Am. Chem. Soc.*, 2018, **140**, 7904–7912.
- M. Campanelli, T. Del Giacco, F. De Angelis, E. Mosconi, M. Taddei, F. Marmottini, R. D'Amato and F. Costantino, *ACS Appl. Mater. Interfaces*, 2019, **11**, 45031–45037.
- L. Luo, L. Huang, X. Liu, W. Zhang, X. Yao, L. Dou, X. Zhang, Y. Nian, J. Sun and J. Wang, *Inorg. Chem.*, 2019, **58**, 11382–11388.



11 Y. Zhang, X. Zeng, X. Jiang, H. Chen and Z. Long, *Microchem. J.*, 2019, **149**, 103967.

12 M. C. Das, S. K. Ghosh, E. C. Sañudo and P. K. Bharadwaj, *Dalton Trans.*, 2009, 1644–1658.

13 O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, *J. Appl. Crystallogr.*, 2009, **42**, 339–341.

14 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Found. Crystallogr.*, 2008, **64**, 112–122.

15 G. M. Sheldrick, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 3–8.

16 (a) T. Hasegawa, S. M. F. Shahed, Y. Sainoo, A. Beniya, N. Isomura, Y. Watanabe and T. Komeda, *J. Chem. Phys.*, 2014, **140**, 044711; (b) M. N. Revoy, R. W. J. Scott and A. P. Grosvenor, *J. Phys. Chem. C*, 2013, **117**, 10095–10105; (c) E. Paparazzo, *Mater. Res. Bull.*, 2011, **46**, 323–326.

17 (a) S. K. Ghosh and P. K. Bharadwaj, *Eur. J. Inorg. Chem.*, 2005, **2005**, 4886–4889; (b) H.-L. Gao, L. Yi, B. Ding, H.-S. Wang, P. Cheng, D.-Z. Liao and S.-P. Yan, *Inorg. Chem.*, 2006, **45**, 481–483.

18 C.-Q. Zhao, Y.-M. Ou, Z.-Y. Zhao, X.-J. Yin and Y.-M. Jiang, *Z. Kristallogr.*, 2013, **228**, 3–4.

19 (a) P. M. Forster, A. R. Burbank, C. Livage, G. Férey and A. K. Cheetham, *Chem. Commun.*, 2004, 368–369; (b) F. Millange, M. I. Medina, N. Guillou, G. Férey, K. M. Golden and R. I. Walton, *Angew. Chem., Int. Ed.*, 2010, **49**, 763–766; (c) H. H.-M. Yeung, Y. Wu, S. Henke, A. K. Cheetham, D. O'Hare and R. I. Walton, *Angew. Chem., Int. Ed.*, 2016, **55**, 2012–2016.

20 I. D. Brown, *The Chemical Bond in Inorganic Chemistry: The Bond Valence Model*, Oxford University Press, Oxford, 2002.

21 T. Rhauderwiek, N. Heidenreich, H. Reinsch, S. Øien-Ødegaard, K. A. Lomachenko, U. Rütt, A. V. Soldatov, K. P. Lillerud and N. Stock, *Cryst. Growth Des.*, 2017, **17**, 3462–3474.

22 S. Sharif, I. U. Khan, S. Zaheer and S. W. Ng, *Acta Crystallogr., Sect. E: Struct. Rep. Online*, 2012, **68**, m624–m625.

23 C.-J. Li, M.-X. Peng, J.-D. Leng, M.-M. Yang, Z. Lin and M.-L. Tong, *CrystEngComm*, 2008, **10**, 1645–1652.

24 I. U. Khan, S. Sharif and O. Sahin, *J. Coord. Chem.*, 2013, **66**, 3113–3125.

25 Y. Xiong, S. Chen, F. Ye, L. Su, C. Zhang, S. Shen and S. Zhao, *Chem. Commun.*, 2015, **51**, 4635–4638.

26 (a) Y. Sha, I. Mathew, Q. Cui, M. Clay, F. Gao, X. J. Zhang and Z. Gu, *Chemosphere*, 2016, **144**, 1530–1535; (b) Z. T. Ildefonso, P. B. José de Jesús, T. L. Celeste Yunueth, L. R. Luis, M. L. Maria Luisa and M. V. Yunny, *RSC Adv.*, 2016, **6**, 76167–76173.

27 (a) N. El-Wakiel and H. El-Ghamry, *Int. J. Chem. Kinet.*, 2017, **49**, 464–476; (b) D. Zhao and C. Cai, *Dyes Pigm.*, 2021, **185**, 108957.

28 X. Tao, W. Cong, L. Huang and D. Xu, *J. Alloys Compd.*, 2019, **805**, 1060–1070.

29 S. S. F. Carvalho, A. C. C. Rodrigues, J. F. Lima and N. M. F. Carvalho, *Inorg. Chim. Acta*, 2020, **512**, 119924.

30 R. Paz, H. Viltres, N. K. Gupta and C. Leyva, *J. Mol. Liq.*, 2021, **337**, 116578.

31 S. Zhao, S. Li, Y. Long, X. Shen, Z. Zhao, Q. Wei, S. Wang, Z. Zhang, X. Zhang and Z. Zhang, *Chemosphere*, 2021, **280**, 130637.

