


 Cite this: *Phys. Chem. Chem. Phys.*,
 2022, 24, 17014

 Received 23rd June 2022,
 Accepted 29th June 2022

DOI: 10.1039/d2cp02864f

rsc.li/pccp

Exploration and validation of force field design protocols through QM-to-MM mapping†

 Chris Ringrose,^a Joshua T. Horton,^a Lee-Ping Wang^b and Daniel J. Cole^{a*}

The scale of the parameter optimisation problem in traditional molecular mechanics force field construction means that design of a new force field is a long process, and sub-optimal choices made in the early stages can persist for many generations. We hypothesise that careful use of quantum mechanics to inform molecular mechanics parameter derivation (QM-to-MM mapping) should be used to significantly reduce the number of parameters that require fitting to experiment and increase the pace of force field development. Here, we design and train a collection of 15 new protocols for small, organic molecule force field derivation, and test their accuracy against experimental liquid properties. Our best performing model has only seven fitting parameters, yet achieves mean unsigned errors of just 0.031 g cm⁻³ and 0.69 kcal mol⁻¹ in liquid densities and heats of vaporisation, compared to experiment. The software required to derive the designed force fields is freely available at <https://github.com/qubekit/QUBEKit>.

1 Introduction

Classical molecular mechanics force fields are widely used approximations to recover the energy and forces of an atomistic system as a function of its nuclear coordinates.^{1,2} Force fields are an invaluable companion to quantum mechanical modelling in cases where the number of atoms or the time scale to be modelled would be otherwise restricted. Such techniques have seen applications in, for example, the modelling of battery materials,^{3,4} organic light-emitting diodes,^{5,6} crystal structure prediction,⁷ and particularly in computer-aided drug design where they form the basis of molecular dynamics, docking and free energy simulations for calculating protein–ligand binding affinity.^{8,9}

Typically, for modelling organic molecules in biology and chemistry, the force field is a sum of bonded (including harmonic bond-stretching and angle-bending terms, and anharmonic 4-body torsion potentials) and non-bonded (including Coulombic and Lennard-Jones interactions) terms. Small molecule force fields of this form include GAFF,¹⁰ CGenFF,¹¹ OPLS,¹² and the Open Force Field 1.0.0 ('Parsley')

force field.¹³ In general, the bonded and charge parameters of the force field may be fit to quantum mechanical potential energy surfaces and electrostatic potentials, respectively. On the other hand, the Lennard-Jones parameters are nearly always fit to experimental pure liquid properties.^{12,14} To be broadly effective, small molecule force fields require accurate parameterisation of each of the above terms for all chemical space. While recent trends have seen parameters fit to larger datasets, complete coverage is challenging. Numerous studies, such as the SAMPL blind challenges¹⁵ have shown that there is plenty of room for improvement when it comes to predictive molecular modelling with force fields.

In contrast to the empirical molecular mechanics force fields common in biological modelling, interatomic potentials may instead be derived directly from quantum mechanics (QM). For example, intermolecular perturbation theory may be used to separate the full QM interaction energy into physically motivated components,¹⁶ or machine learning based potentials may be trained on QM energies and forces.¹⁷ These methods are attractive because they remove the empiricism of the force field approach, but they are computationally more expensive, and unless they fully incorporate, for example, many body and quantum nuclear effects they are unlikely to reproduce condensed phase observables with sufficient accuracy.

QM-to-MM mapping reduces size of parameter search space

Between the empiricism of typical classical force fields and the accurate, yet expensive, *ab initio* derived force fields, increasing attention is being drawn to mapping physically motivated parameters from QM into simple MM functional forms. By deriving

^a School of Natural and Environmental Sciences, Newcastle University, Newcastle upon Tyne NE1 7RU, UK. E-mail: daniel.cole@ncl.ac.uk

^b Department of Chemistry, The University of California at Davis, Davis, California 95616, USA

† Electronic supplementary information (ESI) available: Lennard-Jones derivations, virtual site protocols, additional methods, images of training/test sets and supporting results (PDF). Training tutorial, example ForceBalance input files, raw data and all train/test force field files (ZIP). See DOI: <https://doi.org/10.1039/d2cp02864f>



bespoke force field parameters for the molecule under study directly from QM, the number of transferable, empirical parameters to be fit is significantly reduced. By retaining common MM functional forms, the potentials may readily be implemented in widely used MM software and used in, for example, free energy calculations. Bespoke intramolecular bond, angle and, particularly, torsion parameters may be readily derived from a small number of QM calculations either by fitting to Hessian matrices^{18–21} or potential energy surface scans.^{22,23} Atom-centred partial charges can be routinely derived from either semi-empirical calculations,^{24,25} or QM electrostatic potential fitting,^{26,27} or atoms-in-molecule electron density partitioning.^{28–31}

Less common, but perhaps most interestingly, is the possibility of using atoms-in-molecule electron density partitioning to derive other components of the non-bonded interaction from QM, in particular dispersion coefficients (C_6 ,^{32–35} C_8 ,^{36,37}...), atomic polarisabilities,^{37,38} and off-centre charges to model electronic anisotropy.^{33,34} For example, Visscher and Geerke have derived a polarisable force field model, with a higher order dispersion term derived from iterative Hirshfeld atoms-in-molecule analysis, and applied it to small-molecule amino acid analogues.³⁹ Of the 138 nonbonded parameters in their model, 132 are determined from QM, leaving just six to be fit to experiment. Kantonen *et al.*⁴⁰ employ a mapping of information from minimal basis iterative stockholder (MBIS) atomic electron densities (in particular the decay constants of the electron densities), to derive atom-specific Lennard-Jones parameters. The mapping parameters (two per element) are fit to experimental liquid properties using the ForceBalance software.⁴¹

Our own work has focused on the development of a QUantum mechanical BEspoke (QUBE) force field and associated toolkit (QUBEKit),³⁴ built on QM-to-MM parameter mapping. QUBE bond and angle force field parameters are derived from the QM Hessian matrix of the molecule under study, using the modified Seminario method.⁴² Atomic partial charges are computed from the density derived electrostatic and chemical (DDEC) partitioned atomic electron densities.^{43,44} The Tkatchenko–Scheffler method³² is used to derive C_6 parameters from the same atomic electron densities, and the repulsive part of the Lennard-Jones potential is derived from atoms-in-molecule atomic radii.³³ Once all other parameters are in place, flexible torsion parameters can be fit to QM dihedral scans using interfaces between QUBEKit and external QM and MM software packages.³⁴ A small number of mapping parameters (in this case free atom radii for use in the derivation of Lennard-Jones parameters) is used to ensure accuracy of condensed phase properties,³⁴ and the resulting force fields have also been shown to perform well in the calculation of protein–ligand binding free energies.^{45–47}

Force field design choices can be tested and optimised

The above mentioned force fields derived from QM-to-MM parameter mappings have a simple functional form, are straightforward to derive from a small number of simple QM calculations, and are bespoke to the system under study. However, despite the advantages, there are still multiple design

choices that must be made when constructing a QM-derived force field. These range from the choice of functional form to use for the final force field, to the choice of underlying QM method to compute the electron density, how to partition it between atoms, and how to map the partitioned density to force field parameters. These choices may be regarded as hyperparameters of the model.⁴⁸ Drawing on lessons from the machine learning field, separate models for each hyperparameter should be systematically trained and tested. However, these hyperparameters are often ‘baked in’ to the design of the force field, and correcting known problems can be challenging.⁴⁹ Until recently, the manual training of force field parameters for a given set of hyperparameters (for example, Lennard-Jones parameters for a given charge model) would be extremely time-consuming. However, allied with improvements in data collection and retrieval,⁵⁰ advances in automated parameter fitting to experimental liquid properties now make this possible.^{41,51,52} Most notably, the ForceBalance software enables reproducible and automated force field parameterisation against a set of target data that can include either QM or experimental input data. In the context of the current work, ForceBalance was used to tune the QM-to-MM mapping parameters against experimental liquid data in the aforementioned study by Kantonen *et al.*,⁴⁰ and was also used to rapidly re-fit Lennard-Jones parameters for a range of proposed implicit solvent models (hyperparameters) in the development of the RESP2 method.²⁷

An automated toolkit for force field design, training and testing

Until now, the systematic testing of design choices in MM force fields has been rather limited, and it is almost impossible for individual users to undertake this task due to the scale of the parameter optimisation problem. In what follows, we describe our interface between the QUBEKit and ForceBalance software packages. This open source (including all dependencies) software workflow allows users to make a choice of force field hyperparameters, train the model against experimental liquid properties, and unambiguously test the resulting accuracy. We provide improved protocols for deriving the positions and magnitudes of off-site charges (virtual sites) to model anisotropic electron density from the output of atoms-in-molecule electron density partitioning calculations, as well as tools for including them in torsion parameter fitting. Using a train/test split of 15/51 small, organic molecules, we develop 15 force field protocols that aim to test the accuracy of different choices of (i) underlying QM method and basis set, (ii) atoms-in-molecule electron density partitioning method, (iii) implicit solvent model parameters (used to pre-polarise charges for use in the condensed phase), (iv) mapping of atomic electron densities to LJ parameters, (v) assignment of Lennard-Jones parameters to polar hydrogen atoms, and (vi) use of virtual sites to model anisotropic electron density. We show that a wide range of different choices can be made in the force field design process, some of which do not affect the overall accuracy, and some of which show marked improvements. For our best performing protocol, we provide all of the parameters (including those for halogens and sulphur) required for users to derive their own force fields using the QUBEKit software. With



these developments comes the potential for future rapid design of next-generation force fields.

2 Methods

2.1 QUBEKit workflow

QUBEKit is a largely Python 3.6+ based force field derivation toolkit for mac and Linux operating systems. By combining multiple open-source bioinformatics and quantum chemistry packages into a single workflow, it is possible to generate bespoke force fields for molecular dynamics simulations (Fig. 1(a)). QUBEKit can be run using only Anaconda or PyPI installable open-source packages, with the option to use Gaussian⁵³ for QM and torsion optimisations, and the fortran package Chargemol^{29,30} for atoms-in-molecule analysis.

The general workflow of QUBEKit begins with generating a set of initial parameters for an OPLS-style force field:

$$\begin{aligned}
 U = & \sum_{\text{Bonds}} \frac{k_r}{2} (r - r_0)^2 + \sum_{\text{Angles}} \frac{k_\theta}{2} (\theta - \theta_0)^2 + \sum_{\text{Dihedrals}} \frac{V_1}{2} (1 + \cos(\phi)) \\
 & + \frac{V_2}{2} (1 - \cos(2\phi)) + \frac{V_3}{2} (1 + \cos(3\phi)) + \frac{V_4}{2} (1 - \cos(4\phi)) \\
 & + \sum_{\text{Pairs}} \frac{q_i q_j}{r_{ij}} + 4\epsilon_{ij} \left(\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6 \right)
 \end{aligned}
 \quad (1)$$

where k_r , r_0 , k_θ , θ_0 , V_{1-4} , q_i , ϵ_{ij} and σ_{ij} are force field parameters to be derived. QUBEKit can take as input most industry-standard molecule formats, such as a SMILES string, or pdb or mol2 file. Initial parameters may be generated from the Open Force Field toolkit,¹³ Antechamber,⁵⁴ or a supplied XML file. From there, multiple conformers of the molecule coordinates are generated using the ETKDG⁵⁵ method in RDKit,⁵⁶ and used in a fast but basic optimisation *via* TorchANI,⁵⁷ XTB⁵⁸ or OpenMM.⁵⁹ Of these conformers, the lowest energy result is taken to the QM optimisation stage. The QM structural optimisation is performed using either PSI4⁶⁰ or Gaussian,⁵³ both run through QCEngine⁶¹ with an ultra-fine grid. If the first, lowest energy conformer fails to optimise after 50 iterations, the next conformer is passed instead. The QM optimised coordinates are used as input for an electron density calculation, a Hessian matrix calculation and optional 1D torsiondrive calculation(s) (Fig. 1(a)). The Hessian matrix is used to calculate all bond and angle parameters (k_r , r_0 , k_θ , θ_0) *via* the modified Seminario method.⁴²

QUBE uses atoms-in-molecule (AIM) electron density partitioning to obtain all non-bonded parameters of the force field. AIM methods partition the total electron density $n(\mathbf{r})$ into overlapping atomic densities $n_i(\mathbf{r})$ according to:

$$n_i(\mathbf{r}) = \frac{w_i(\mathbf{r})}{\sum_k w_k(\mathbf{r})} n(\mathbf{r}) \quad (2)$$

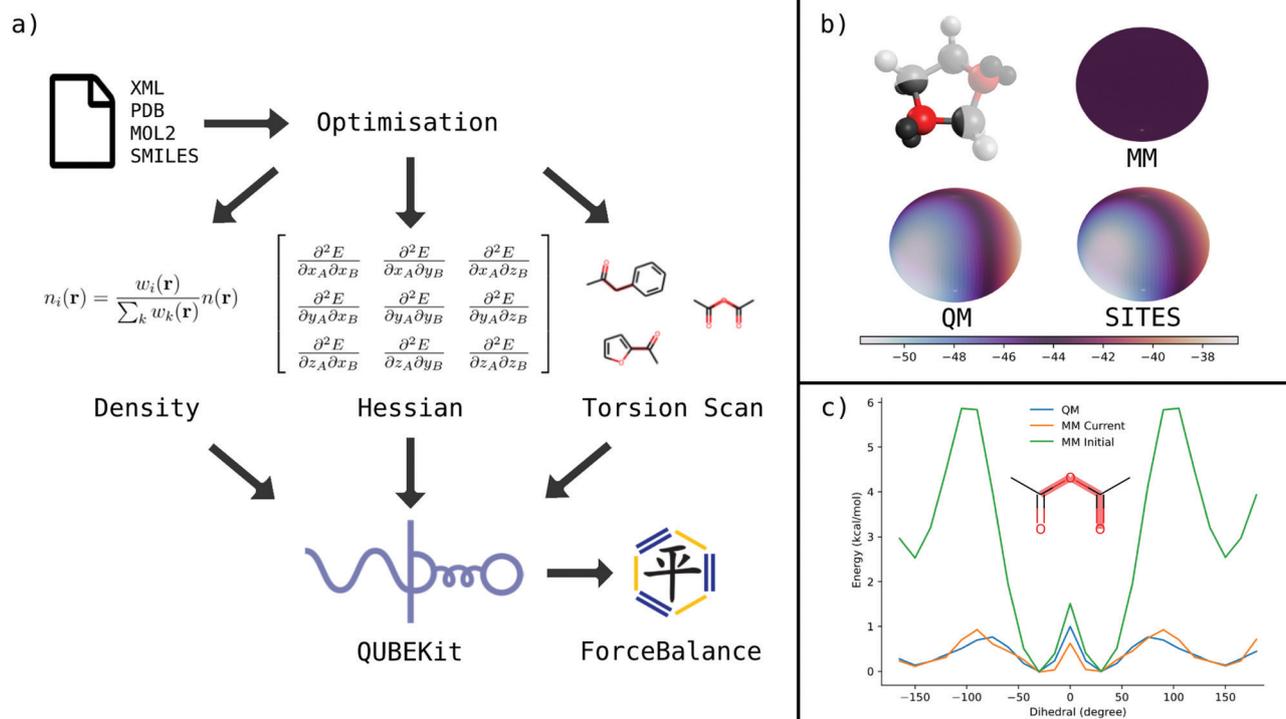


Fig. 1 (a) QUBEKit workflow. The molecule under study is input and structurally optimised. The total electron density is partitioned into atomic contributions, while Hessian matrices and optional torsion scans are computed. QM measurements are mapped to MM force fields using the QUBEKit software, and (optionally) mapping parameters are optimised using ForceBalance. (b) Virtual site derivation for 1,3-dioxolane. (top right) ESP surface plot (at $1.4 \times$ the van der Waals radius) using a single atom-centred point charge (kcal mol^{-1}), (bottom left) the QM ESP up to quadrupole order, (bottom right) the MM ESP with the addition of two virtual sites (as shown in top left). (c) Example torsion scans before (green) and after (orange) parameter optimisation.



where the form of the weights, $w_i(\mathbf{r})$, are determined by the choice of AIM method. Common choices include (iterative) Hirshfeld,³¹ MBIS²⁸ and density derived electrostatic and chemical (DDEC)^{29,30} partitioning. Following AIM partitioning, atom-centred partial charges are simply assigned by integrating the atomic electron densities:

$$q_i = z_i - \int n_i(\mathbf{r}) d^3\mathbf{r} \quad (3)$$

where z_i is the nuclear charge. It is well-known that partial charges in non-polarisable force fields need to be scaled in some way to account for polarisation in an effective manner in condensed phase simulations. Common approaches include use of bond order corrections or charge scaling factors,^{24,25} but we prefer to use an implicit solvent model with a dielectric constant intermediate between the gas and water phases.^{27,33,62} The final charges thus depend on the choice of underlying QM method used to compute $n(\mathbf{r})$, the choice of AIM method, and the choice of implicit solvent model and parameters. Previously we computed the electron density using the linear-scaling DFT code, ONETEP,⁶³ using the PBE exchange–correlation functional and a non-orthogonal generalised Wannier function basis set.³⁴ A multigrid Poisson solver⁶⁴ was used with a dielectric of 4 to pre-polarise the charges, and an implementation of the DDEC AIM method in ONETEP (similar to DDEC3⁴⁴) was employed to partition the electron density. As discussed in the Introduction, we now have the infrastructure in place to systematically test the effects of these hyperparameter choices. Table 1 summarises the alternative model protocols that will be investigated here to test the effect of the underlying QM method (models **1a** and **1b**), the implicit solvent parameters (models **2a–2c**) and AIM partitioning scheme (models **3a** and **3b**) on the force field parameters, and physical properties.

In the original QUBE force field, the non-bonded part of the force field includes the Lennard-Jones interaction:

$$V_{\text{LJ}}(r_{ij}) = \frac{A_{ij}}{r_{ij}^{12}} - \frac{B_{ij}}{r_{ij}^6} \quad (4)$$

between pairs of atoms i, j . The dispersion coefficient, B_i is derived for the atom in the molecule *via* the Tkatchenko–Scheffler (TS) relation:³²

$$B_i = \left(\frac{V_i^{\text{AIM}}}{V_i^{\text{free}}} \right)^2 B_i^{\text{free}} \quad (5)$$

where V_i^{AIM} is the third radial moment of the partitioned atomic electron density:

$$V_i^{\text{AIM}} = \int r^3 n_i(\mathbf{r}) d^3\mathbf{r} \quad (6)$$

The corresponding quantities B_i^{free} and V_i^{free} may be computed and tabulated for free atoms in a vacuum (Table S1, ESI†). To ensure that the Lennard-Jones potential has a minimum at the effective van der Waals radius of the atom-in-molecule, the coefficient of the repulsive term may be approximated *via*:³³

$$A_i = \frac{1}{2} B_i (2R_i^{\text{AIM}})^6 \quad (7)$$

where the AIM radius, R_i^{AIM} , is again found by re-scaling a reference free atom radius:

$$R_i^{\text{AIM}} = \left(\frac{V_i^{\text{AIM}}}{V_i^{\text{free}}} \right)^{1/3} R_i^{\text{free}} \quad (8)$$

The R_i^{free} are free parameters to be fit to experiment, which we have found to be crucial if the force field is to reproduce condensed phase properties. Previously, we have fit these *via* parameter scans to liquid properties (densities and heats of vaporisation).³³ However, as discussed in the Introduction, a new set of parameters is required for each set of model hyperparameters, and so we employ here a more automated approach as set out later.

Transforming the A_i and B_i parameters to the σ_i and ϵ_i parameters of eqn (1), we obtain (a full derivation is given in ESI,† S1.1):

$$\sigma_i = 2^{5/6} \left(\frac{V_i^{\text{AIM}}}{V_i^{\text{free}}} \right)^{1/3} R_i^{\text{free}} \quad (9)$$

Table 1 Summary of changes across force field models. Model **0** is designated as the default protocol, and changes from the default in subsequent models are highlighted in bold. Training set mean unsigned error (MUE) in density (g cm^{-3}) and heat of vaporisation (kcal mol^{-1}), relative to experiment, are displayed for each force field protocol

Model	QM method	Solvent	V-Sites	AIM	LJ	MUE ρ	MUE ΔH_{vap}
0	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	No	DDEC6	TS	0.0274	0.766
1a	B3LYP-D3(BJ)/DZVP	IPCM, $\epsilon = 4$	No	DDEC6	TS	0.0271	0.726
1b	HF/6-31G(d)	None	No	DDEC6	TS	0.0248	0.731
2a	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 2$	No	DDEC6	TS	0.0296	0.772
2b	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 10$	No	DDEC6	TS	0.0237	0.783
2c	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 20$	No	DDEC6	TS	0.0285	0.678
3a	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	No	DDEC3	TS	0.0213	0.475
3b	B3LYP-D3(BJ)/DZVP	Chloroform	No	MBIS	TS	0.0159	0.578
4a	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	No	DDEC6	HO	0.0235	0.896
4b	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	No	DDEC6	α, β	0.0206	0.587
5a	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	Yes	DDEC6	TS	0.0190	0.441
5b	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	Yes	DDEC6	α, β	0.0209	0.244
5c	ω B97X-D/6-311++G(d,p)	IPCM, $\epsilon = 4$	Yes	DDEC3	TS	0.0163	0.357
5d	B3LYP-D3(BJ)/DZVP	IPCM, $\epsilon = 4$	Yes	DDEC6	TS	0.0141	0.450
5e	B3LYP-D3(BJ)/DZVP	Chloroform	Yes	MBIS	α, β	0.0175	0.480



and:

$$\varepsilon_i = \frac{B_i^{\text{free}}}{2(2R_i^{\text{free}})^6} \quad (10)$$

Throughout this study, the Lorentz–Berthelot combination rules are used to derive ε_{ij} and σ_{ij} from the corresponding atomic quantities. Eqn (9) results in atoms with more diffuse electron density having a larger Lennard-Jones σ_i parameter but, as correctly pointed out previously,⁴⁰ all atoms of the same element have the same Lennard-Jones well depth (eqn (10)). In that same study, the authors proposed that each atom should have a unique effective ionisation energy, derived from the partitioned atomic electron density, which affects the scaling relationship in eqn (5).⁴⁰ An additional consideration is that the dispersion coefficient (B_i) should be viewed as an effective interaction, taking into account not only the dipole-dipole, but also higher-order (dipole–quadrupole *etc.*) contributions to dispersion. It has been shown that physics-based derivations of B_i tend to therefore be lower than the effective coefficients in common MM force fields.^{35,45} To take both of these factors into account, we test here the effect of introducing additional flexibility in the definition of the dispersion coefficient:

$$B_i = \alpha \left(\frac{V_i^{\text{AIM}}}{V_i^{\text{free}}} \right)^{2+\beta} B_i^{\text{free}} \quad (11)$$

where α and β are global fitting parameters. This change has no effect on σ_i , but ε_i is now dependent on the diffuseness of the atomic electron density, measured by V_i^{AIM} (a full derivation is given in the ESI,[†] S1.2):

$$\varepsilon_i = \frac{\alpha \left(\frac{V_i^{\text{AIM}}}{V_i^{\text{free}}} \right)^\beta B_i^{\text{free}}}{2(2R_i^{\text{free}})^6} \quad (12)$$

Note that β can be positive or negative. Thus, as summarised in Table 1, we can set $\alpha = 1$ and $\beta = 0$ to retain the original QUBE force field model (*e.g.* model **0**), or optimise α and β alongside R_i^{free} (*e.g.* model **4b**). The effects of these choices are discussed later.

An additional choice in force field design is whether to include Lennard-Jones parameters on polar hydrogen atoms, or to effectively redistribute them onto the neighbouring heavy atom³³ (see ESI,[†] S1.3). Different choices are made for example in the design of the TIP3P and CHARMM modified water model,⁶⁵ while there is precedent for improved agreement with liquid data using Lennard-Jones parameters on polar hydrogen atoms.⁶⁶ While separate mapping parameters have been previously proposed for polar and non-polar hydrogen atoms, the effect on accuracy has not been directly explored,⁴⁰ and so we add this comparison to our list of model protocols to be tested (Table 1, model **4a**).

A final consideration to be made when deriving the non-bonded parameter set is how to treat atoms with significant anisotropy in their electron density, such that an atom-centred point charge gives a poor approximation to the full QM electrostatic potential at the surface of the molecule. Such situations

are commonly treated in MM force fields using an off-centre charge (or “virtual site”), as used in, for example, the TIP4P water model⁶⁷ and various treatments of lone-pairs and σ -holes in common force fields.^{68–70} In keeping with our general force field philosophy, we previously proposed a method to derive the positions and magnitudes of off-site charges, where required, directly from the partitioned QM electron density. Full details are given elsewhere,³⁴ but in brief the charges are positioned to minimise the quantity:

$$\Phi = \frac{1}{n} \sum_{i=1}^n \left| V_{\text{QM}}^i - V_{\text{MM}}^i \right| \quad (13)$$

where V_{QM}^i is the electrostatic potential generated by the partitioned atomic electron density at a sample point i , V_{MM}^i is the corresponding quantity calculated using the point charge model, including any off-centre charge(s), and n is the total number of sample points (located between 1.4 to 2.0 times the van der Waals radius of the atom). By using the atomic, rather than the molecular, electron density, the method scales easily to larger molecules. To limit the search space, off-site charges are restricted to positions determined by the symmetry of the atom’s bonding environment.³⁴ For example, for halogens bonded to a single atom, the search direction is along the bond vector, and for oxygen bonded to two neighbours, the search direction is along the bisector of the two bonds. Similar arguments can be made for positioning two virtual sites.³⁴

A similar protocol is followed in the current work. Key differences are discussed here and full implementation details are provided in the ESI,[†] Section S2. It is desirable to perform the optimisation of eqn (13) directly in QUBEKit, but storage and imports of the full atomic electron density is inefficient. Instead, V_{QM}^i is now reconstructed from the QM atomic multipole moments, up to quadrupole order, which are readily extracted from either Chgemo (using the DDEC AIM method)^{29,30} or the PSI4 software (using the MBIS AIM method).⁶⁰ For atoms with isotropic electron density, the atomic multipoles are low, and the QM electrostatic potential is well-described by an atom-centred point charge (monopole). For atoms with anisotropic electron density (defined here as $\Phi > 1$ kcal mol^{−1}), eqn (13) is minimised with respect to the virtual site charge and position, using the Scipy python package.

Fig. 1(b) shows example surface plots of the electrostatic potential energy (ESP) at 1.4× the van der Waals surface of an oxygen atom in 1,3-dioxolane. The bottom-left plot shows the QM-calculated ESP, up to quadrupole order. The top-right plot shows the ESP using an atom-centred monopole, which simply gives a uniform, averaged ESP over the entire surface of the atom. The difference between these two plots at every point allows the calculation of the error function, eqn (13). The bottom-right plot shows an approximation of the QM ESP, constructed with the addition of just two virtual sites, which are able to recover the expected anisotropy in the ESP. Table 1 shows the experiments performed in what follows to test the effect of including virtual sites on our force field model accuracy (models **5a–5e**).



In previous works, new non-bonded force field parameters tend to be investigated in conjunction with the bonded parameters of standard transferable force fields, which may lead to incompatibility given the close inter-dependency between parameter types. Here, the benefit of assembling the force fields using the QUBEKit package, is that torsion parameters for rotatable bonds may be re-fit in a simple, automated manner for compatibility with the rest of the bonded and non-bonded parameters. Using a new interface between QUBEKit, ForceBalance⁴¹ and TorsionDrive,⁷¹ dihedral parameters of any freely rotatable bonds were optimised *via* a least squares minimisation of the root mean square error (RMSE) between QM and MM torsional potential energy surfaces. We use the same procedure for parameter fitting as the recent Open Force Field Parsley force field,¹³ and full details are provided in ESI,† Section S3.1. A new feature in this work is that we have modified the ForceBalance code to allow parameter fitting in the presence of local coordinate virtual sites. These virtual sites then inherit their exceptions and exclusions from the parent atom meaning they interact with the same rules as the parent. The code may be obtained from conda-forge or github (<https://github.com/leeping/forcebalance>), and is available from version 1.9.0 onward. Fig. 1(c) demonstrates an example of bespoke dihedral parameter fitting for the molecule acetic anhydride, which has two virtual sites on the central oxygen atom. With the initial parameter set (which is incompatible with the new Lennard-Jones parameters and virtual sites), the RMS error is more than 3 kcal mol⁻¹, but following fitting it falls to just 0.12 kcal mol⁻¹. In fact, the average RMSE across all molecules in the 51 molecule test set (a total of 117 scans) after fitting was just 0.13 kcal mol⁻¹.

2.2 ForceBalance workflow

As outlined in Fig. 1(a), for each choice of force field design protocol (Table 1), the ForceBalance software⁴¹ was employed to optimise the set of fitting parameters. These parameters were the R_i^{free} parameters of H, C, N and O (eqn (8)), and optionally the dispersion rescaling parameters α and β (eqn (11)). Where Lennard-Jones parameters were included on polar hydrogen atoms, it was found that a separate R_i^{free} parameter was required for these atoms. Optimisation was performed against the experimental liquid densities and enthalpies of vaporisation of a training set of 15 molecules. The training set comprised molecules containing H, C, N and O only, and a range of functional groups (ESI,† Section S4). It was identical to that used in the training of the RESP2 charge fitting procedure,²⁷ except that one molecule (N1(C)CCOCC1) was replaced by CN(C)C=O to ensure that sampling complex potential energy surfaces did not influence the fit. For two of the best performing models, a further ten molecules containing halogens (F, Cl and Br) and sulphur atoms were added to the training set,⁵¹ and the fitting parameters were co-optimised for the combined set of 25 molecules.

ForceBalance uses the reference data to produce an objective function, which is minimised with a non-linear optimisation algorithm through numerical differentiation of physical properties with respect to the fitting parameters. Further details are given in the ESI,† Section S3.2. To create the inputs required by

ForceBalance, QUBEKit automatically combines the force fields of the individual molecules, along with algebraic expressions that transform the fitting parameters into the σ_i and ϵ_i parameters of the force field (see, for example, eqn (9) and (10)), into a single xml file. The full training sets, experimental data, example ForceBalance input files, and tutorial are provided in the ESI.†

Following the completion of ForceBalance training runs, the final fitting parameters are used to re-calculate the Lennard-Jones and re-fit the torsion parameters in QUBEKit. Training set accuracy was then confirmed by re-computing the liquid properties of the training set molecules using QUBE Bench, which is an interface with OpenMM for the benchmarking of QUBE force fields (ESI,† Section S3.3). QUBE Bench and ForceBalance estimates of force field accuracy (relative to experiment) deviated by just 0.002 g cm⁻³ and 0.11 kcal mol⁻¹, on average, for density and heat of vaporisation calculations, respectively, indicating good convergence of the fit. Finally, four of the most encouraging force field protocols were taken forward to testing. Here, QUBEKit was used to fit force fields for a test set comprising a further 51 molecules taken from a previous study.²⁷ An additional set was built to test parameters for molecules containing halogens and sulphur. The molecules in both test sets are listed in full in ESI,† Section S4. Physical properties were calculated using QUBE Bench, and compared with experiment, to confirm transferability of the designed protocols.

3 Results

3.1 Training set accuracy

Using our new interface between the QUBEKit engine for QM-to-MM force field parameter mapping and the ForceBalance software for parameter tuning, we have trained a total of 15 different force field protocols to investigate the effects of choices made in force field design on model accuracy. Descriptions of the force field protocols are listed in full in Table 1, grouped roughly by the type of hyperparameter being investigated.

The first groupings investigate the choice of underlying QM methods used to compute the total electron density for AIM partitioning and subsequent non-bonded parameter derivation (the same QM method is also used for Hessian matrix calculation and dihedral scans to obtain the bonded parameters). As mentioned, our first implementation of QUBEKit³⁴ interfaced with the ONETEP DFT code,⁶³ which uses the PBE exchange-correlation functional, for non-bonded parameter derivation. However, this new version of QUBEKit is interfaced, *via* QCEngine,⁶¹ with both Gaussian09⁵³ and PSI4,⁶⁰ which gives us access to a wide range of alternative quantum chemistry methods. If using PSI4, the electron density partitioning is performed using MBIS.²⁸ If using Gaussian09, Chargemol is used to calculate the atom centred charges *via* either the DDEC3 or DDEC6 AIM schemes. In all cases, the AIM method returns atom-centred point charges, the third radial moment of the atomic electron density (for Lennard-Jones parameter derivation), and multipole moments of the atomic electron density (for virtual site derivation).



QM method. First, we investigate the use of two relatively high level QM methods. The ω B97X-D/6-311++G(d,p) method is used for bonded parameter derivation in earlier QUBE force fields,^{34,72} as well as the OPLS-AA/M protein force field.⁷³ The B3LYP-D3(BJ)/DZVP method is used for bonded parameter fitting in the Parsley force field¹³ and has been shown to give a good compromise between accuracy and computational expense for gas phase conformational energetics of small organic molecules.⁷⁴ It has been shown that the choice of DFT functional is not too critical in the reproduction of molecular electrostatic potentials, while larger basis sets afford some improvements in accuracy.²⁷ However, relatively little is known about how this choice translates to force field accuracy. Runs **0** and **1a** reveal that this choice is not critical, with nearly identical training set accuracy obtained for both liquid densities (0.027 g cm^{-3}) and heats of vaporisation (0.77 vs. $0.73 \text{ kcal mol}^{-1}$). It should be emphasised that this does not mean that the non-bonded parameters themselves are identical. Table 2 shows the final R_i^{free} parameters after ForceBalance training. There is a notable reduction in the radius of the N atom, using the B3LYP-D3(BJ)/DZVP method, and corresponding increase in both the polar and non-polar H atoms. This backs up the recent assertion that improving the accuracy of atomic charges is unlikely to improve calculation accuracy without a corresponding optimisation of Lennard-Jones parameters.²⁷

For comparison, in run **1b**, we employed the HF/6-31G(d) method, which has been used historically for the fitting of force field ESP charges.²⁶ The argument goes that this method produces a fortuitous over-polarisation of the electron density in the gas phase to yield charges that are suitable for condensed phase modelling, and so we perform these electron density calculations in vacuum (Table 1). The results are very similar to the two higher level QM methods in implicit solvent, but perhaps this is not too surprising, given the success of many force fields that are built from the HF/6-31G(d) method. However, as we will show, we can now do better with improved physics-based protocols.

Implicit solvent model. In previous versions of QUBEKit, we have computed the electron density using a minimal parameter

solvent model, with the solute cavity defined by an isosurface of the electron density.⁶⁴ We argued that a solvent dielectric $\epsilon = 4$ is appropriate, as it leads to charges that are polarised midway between vacuum and a dielectric medium of $\epsilon = 78$ (*i.e.* water).³³ Gaussian09 provides the IPCM implicit solvent model,⁷⁵ which also builds the solute cavity from an isosurface of the electron density, and so we use this model as our default here. Models **2a–2c** compare the effect of tuning the solvent dielectric in the range $\epsilon = 2–20$. Again, very little change in force field accuracy is observed, with density errors in the range $0.02–0.03 \text{ g cm}^{-3}$ and heat of vaporisation errors in the range $0.6–0.8 \text{ kcal mol}^{-1}$. This is broadly in agreement with previous observations,²⁷ which showed a strong dependence of condensed phase properties on the polarity of the charge model for a fixed set of Lennard-Jones parameters, but much less sensitivity when the Lennard-Jones parameters are tuned for consistency with the charges. Table 2 shows that the R_i^{free} parameters in run **2a** tend to be lower than the others, which indicates that the minimum of the Lennard-Jones interaction should be closer to the atom for the less-polarised charge model. Compared to vacuum, we found that molecular dipole moments are scaled by a factor of $1.10 \times (\epsilon = 2)$ and $1.22 \times (\epsilon = 20)$, on average, indicating that a suitable range of condensed phase polarisation has been captured in these force field models by tuning the dielectric of the background implicit solvent (Fig. S5, ESI†).

AIM partitioning method. Our previous version of QUBEKit relied on an implementation of the DDEC AIM approach in the ONETEP DFT code.⁴⁴ The DDEC AIM method is described elsewhere,²⁹ but in short it aims to construct the optimal weighting factors of eqn (2) that result in approximately spherical atomic electron densities, so that its multipole expansion converges rapidly, whilst ensuring that the resulting charges are chemically reasonable. Alongside the observation that DDEC charges show excellent transferability between different conformations of the same molecule,³⁰ these properties make the DDEC AIM approach very promising for flexible force field design.

In the previous sections, we employed the latest DDEC6 AIM approach for the first time in organic molecule force field design, through the interface between Gaussian09 and Chargemol. In model **3a**, we investigate the effect of switching to the older DDEC3 approach. The accuracy is actually slightly better for DDEC3, with a decrease in both density and heat of vaporisation errors, compared to the baseline model **0**. DDEC6 has several methodological improvements, which are summarised elsewhere²⁹ and have been shown to result in more robust convergence, lower computation times, lower ESP errors, and higher transferability, when compared to DDEC3.^{29,30} However, most of these advantages will not be apparent in the small, relatively rigid, organic molecules, which lack buried atoms and are simulated here under standard conditions.

MBIS is similar in idea to the Hirshfeld family of AIM methods.²⁸ In this method the total electron density is partitioned onto a minimal set of *s*-type Slater functions, whose parameters are fit to the input molecular electron density by minimising the Kullback–Leibler divergence. Like DDEC, it has been shown that the MBIS-derived charges reproduce the QM ESP to good accuracy

Table 2 Values of the fitting parameters following ForceBalance optimisation for each of the force field protocols. The R_i^{free} for each element are in Å, and α and β are dimensionless

Model	C	N	O	H	Polar H	α	β
0	2.008	1.765	1.499	1.738	1.083	—	—
1a	1.999	1.740	1.489	1.752	1.111	—	—
1b	2.042	1.676	1.501	1.737	1.218	—	—
2a	2.004	1.708	1.464	1.744	1.107	—	—
2b	2.026	1.751	1.521	1.724	1.119	—	—
2c	2.025	1.756	1.503	1.733	1.132	—	—
3a	2.051	1.740	1.590	1.670	1.126	—	—
3b	2.068	1.681	1.599	1.753	1.404	—	—
4a	2.021	1.604	1.550	1.719	—	—	—
4b	2.074	1.742	1.481	1.760	1.154	1.301	0.465
5a	1.994	1.706	1.558	1.738	1.279	—	—
5b	2.035	1.722	1.574	1.731	1.294	1.221	0.489
5c	2.042	1.740	1.630	1.687	1.274	—	—
5d	2.013	1.680	1.558	1.732	1.442	—	—
5e	2.043	1.693	1.680	1.680	1.464	0.999	0.491



and are robust to small conformational changes, while MBIS can also be used to derive Tkatchenko–Scheffler based dispersion coefficients.²⁸ MBIS charges, along with all AIM properties (atomic multipoles up to quadrupole order and radial moments) that we require for force field derivation have recently been implemented in the PSI4 quantum chemistry package,⁶⁰ which we can now access through our latest interface with QCEngine.⁶¹ Model **3b** thus investigates the use of MBIS as the AIM partitioning method in force field derivation. For technical reasons, it was necessary to use the B3LYP exchange-correlation functional, and the IEFPCM implicit solvent model (with a chloroform solvent to mimic a dielectric of approximately 4),⁷⁶ for the underlying QM calculations. Again this combination of methods gave very good accuracy on our training set, with errors in the density and heat of vaporisation of 0.016 g cm^{-3} and $0.58 \text{ kcal mol}^{-1}$, respectively.

Given the expected benefits of DDEC6 in larger, more flexible molecules, we retain this method as our standard approach, but do investigate both DDEC3 and MBIS further in what follows. It is gratifying that a range of AIM methods has been shown to be promising for flexible force field design, and is available through our QUBEKit interface with QCEngine.⁶¹

Lennard-Jones parameters. As discussed in the Methods section, the default QUBE protocol for deriving the attractive part of the Lennard-Jones potential (B_{ij} in eqn (4)) from QM tends to lead to dispersion coefficients that are lower than those used in common effective force fields⁴⁵ and energy well depths that are constant across atoms of the same element.⁴⁰ Model **4b** in Table 1 investigates the effects of relaxing this protocol, by introducing two new global, variable parameters (α and β) into the fit, thus moving away from the default Tkatchenko–Scheffler (TS) approach. As expected, a reduction in training set errors is achieved, relative to the default model **0**, with these extra fitting parameters. Table 2 shows that this is achieved with $\alpha > 1$ and $\beta > 0$, both factors that act to increase the well depth for atoms with more diffuse electron density (eqn (12)). This behaviour is further highlighted in Fig. S6 (ESI[†]), which shows example Lennard-Jones potential energy surfaces for models **0** and **4b**.

Fig. 2 plots the dispersion coefficients (summed over all atoms in the molecule) for all molecules in the larger test set, using the identical Lennard-Jones scheme (from model **5b**, see later), against the corresponding quantities extracted from the Parsley force field. Unlike previous protocols (see our previous work,⁴⁵ and Section S5.3, ESI[†]), there is a very good correlation between the two force fields, albeit with a consistent offset. While the Parsley parameters are extracted from a library, which in turn has been fit to condensed phase properties, and ours are derived from QM calculations specifically for the molecule under question, the convergence of the two approaches is encouraging.

In addition, we investigate the question of whether to include Lennard-Jones parameters on polar hydrogen atoms, or to set them to zero and effectively absorb them onto the neighbouring heavy atom (Section S1.3, ESI[†]).³³ In Table 1, we show that force field protocol **4a**, which has no Lennard-Jones parameters on polar hydrogens (denoted ‘H0’), has a similar

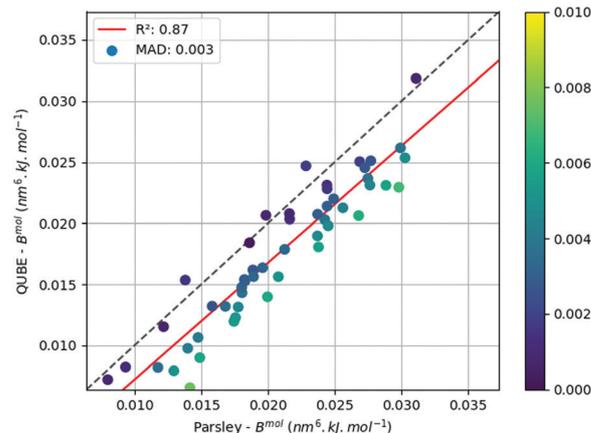


Fig. 2 Correlation between Parsley and QUBE (model **5b**) summed molecular dispersion coefficients for each molecule in the test set

$$\left(B^{\text{mol}} = \sum_{i \in \text{mol}} B_i \right).$$

density error, but higher heat of vaporisation error ($0.90 \text{ kcal mol}^{-1}$) compared to the other protocols. Interestingly, this model has a much lower K_i^{free} for nitrogen than the other models, but the low density error indicates that this does not propagate through to too short hydrogen bonding distances in condensed phase simulations. Nevertheless, we conclude that explicitly including Lennard-Jones parameters on polar hydrogen atoms has accuracy benefits, and we continue to use this approach in the following force field models.

Virtual sites. Fig. 1(c) shows an example benefit of including just a small number of virtual sites on a molecule in terms of the reproduction of the QM electrostatic potential (in this case up to quadrupole order). In fact, considering a larger test set of 51 molecules, we found that 33 molecules had electron density anisotropy above our set threshold (1 kcal mol^{-1} , eqn (13)) for at least one atom. As expected from our previous work,³⁴ these virtual sites were found almost exclusively on oxygen and nitrogen atoms. Using atom-centred point charges only, the average ESP error on these atoms was $1.87 \text{ kcal mol}^{-1}$, which fell to $0.53 \text{ kcal mol}^{-1}$ after virtual site fitting. Further examples are given in the next section and a full list of molecules and ESP errors is given in the ESI[†].

However, a more pertinent question is whether a force field model that more accurately captures the QM electrostatic potential also results in more accurate prediction of condensed phase properties. We have therefore trained a series of force field protocols that include virtual sites in the description of the electrostatics (models **5a–5e**). Table 1 summarises the accuracy on the training set. As opposed to some of the other design choices, we now see significant accuracy gains for these models, with prediction errors in liquid densities in the range $0.014\text{--}0.021 \text{ g cm}^{-3}$ and heats of vaporisation between $0.24\text{--}0.48 \text{ kcal mol}^{-1}$. The lowest density error is observed when virtual sites are used in combination with the B3LYP-D3(BJ)/DZVP QM method (model **5d**), and the lowest energy errors when used with rescaled Lennard-Jones interactions (model **5b**). Although there is no significant trend in the fit R_i^{free} parameters



on the heavy atoms for these force field protocols (Table 2), interestingly there is a consistent shift of the van der Waals radii of the polar hydrogen atoms to higher values (1.27–1.46 Å) when used in combination with virtual sites. This makes intuitive sense since these atoms are likely to be involved in hydrogen bonds with anisotropic polar atoms, like oxygen and nitrogen. The virtual sites will move the centre of electron density closer to the van der Waals surface of the heavy atom, meaning that the Lennard-Jones repulsion should increase to compensate the increase in electrostatic attraction. The improved condensed phase properties seem to indicate that this is a more physically-reasonable balance than the traditional atom-centred point charge approach.

Expanding the training set. For force field models **5b** and **5d**, the training set was further expanded to provide parameters for the halogens, F, Cl and Br, as well as sulphur. A further ten molecules were added to the training set (Fig. S2, ESI†), and all R_i^{free} (and α and β for **5b**) were re-optimised. The new parameter sets are provided in Table S4 (ESI†). Comparing with Table 2, there is little change in the existing parameters upon re-optimisation (maximum change of 0.04 Å in the N fitting parameter in model **5d**) and the new R_i^{free} are physically reasonable given the relative sizes of the atoms. For the expanded training set, the accuracy of model **5b** (MUEs of 0.019 g cm⁻³ and 0.50 kcal mol⁻¹ in density and heat of vapourisation) and **5d** (MUEs of 0.019 g cm⁻³ and 0.55 kcal mol⁻¹) are quite similar to each other. Correlation plots between predicted and experimental condensed phase data are presented in Fig. S11 (ESI†). Overall, there is a small decrease in model accuracy, when compared to Table 1, but we have shown that the force field protocols developed here are also transferable to modelling larger atoms, often with quite anisotropic electron density. On that last point, Fig. S10 (ESI†) shows a selection of virtual sites applied using QUBEKit to the new molecules in the training set. Most are intuitive, such as lone pair positions on S atoms and a σ -hole on Br. Interestingly, in molecules such as chlorofluoromethane, we often see virtual sites positioned along the C–F bond (which acts to significantly decrease the ESP error), and no sites added to Cl atoms (since the atom-centred point charge ESP error is below our threshold). The good accuracy of our model in the condensed phase supports these assignments, and may also help to influence virtual site placements in future transferable force field design.

3.2 Test set accuracy

To test the accuracy of the designed force fields on molecules outside the training set, we ran simulations of the condensed phase properties of a separate test set of 51 molecules containing H, C, N and O atoms only. The force fields brought through for study were models **0**, **1a**, **5b** and **5d** (Table 3). The first two were chosen as our simplest (but also amongst the least accurate) protocols, differing only by the choice of underlying QM method. In both cases, we see a drop in overall accuracy, compared to the training set, which is to be expected for the larger, more complex molecules found in the test set (Fig. S3, ESI†). Again, we do not observe a strong dependence of the results on the choice of QM

method, and so the computationally less expensive model **1a**, based on the B3LYP-D3(BJ)/DZVP method, should make a good compromise force field. For comparison, we have also run the test set using the Open Force Field Parsley force field¹³ using identical protocols. The accuracy in the simulated densities (0.038 g cm⁻³) is similar to our first two protocols, and the error in the heat of vaporisation (1.18 kcal mol⁻¹) is slightly lower. The Parsley force field is based around the AM1-BCC charge scheme, with Lennard-Jones parameters largely retained from the SMIRNOFF99Frosst force field.⁷⁷ Since our initial comparison, Open Force Field have released an updated force field (2.0.0, ‘Sage’) with further refinement of the Lennard-Jones parameters. This force field gives corresponding errors of 0.030 g cm⁻³ and 1.16 kcal mol⁻¹ on our test set, albeit with many more adjustable parameters than we use in our QM-to-MM mapping potentials (the five R_i^{free} parameters in Table 2).

Two additional force field protocols were chosen for further investigation based on low training set errors. Model **5d** uses the B3LYP-D3(BJ)/DZVP method, and virtual sites to model anisotropic electron density. In this case, we see moderate improvement in test set accuracy for both density and heat of vaporisation (Table 3). Model **5b** uses our default quantum chemistry method, virtual sites and re-scaled Lennard-Jones interactions. With approximately the same density error, the error in the heat of vaporisation is significantly reduced to around 0.7 kcal mol⁻¹ (Fig. 3). Only two compounds have heat of vaporisation errors exceeding 2 kcal mol⁻¹, and these are two relatively long chain molecules containing hydroxyl groups (SMILES: OCCCCCO and CC(C)(O)CCC(C)(C)O). Further analysis of models **5b** and **5d** on an additional test set incorporating halogen and sulphur atoms is provided in ESI,† Section S5.5. Overall, the heat of vaporisation accuracy remains close to 1 kcal mol⁻¹, and density errors increase relative to those shown in Table 3 (though still remain lower than Parsley).

It is difficult to pinpoint reasons for improvement in accuracy, since the physical properties are derived from many competing effects. However, Fig. 4 shows some of the force field parameters for molecules that show differences in performance between our best model (**5b**), and Parsley. Fig. 4(a) shows 1,3-benzodioxolane, which has $\Delta H_{\text{vap}}^{\text{exper}} = 13.1$, $\Delta H_{\text{vap}}^{\text{QUBE}} = 13.8$ and $\Delta H_{\text{vap}}^{\text{parsley}} = 15.3$ kcal mol⁻¹. The net charge and Lennard-Jones ϵ parameter are similar for both QUBE and Parsley, but the former has the charge roughly evenly spread over the atom centre and two virtual sites, showing the benefits of modelling anisotropy in the electron density. Fig. 4(b) shows heptane, and in this case

Table 3 Test set accuracy. MUE in density (ρ) and heat of vaporisation (ΔH_{vap}) for 51 molecules in the test set. 95% confidence intervals are included

Force field	MUE ρ (g cm ⁻³)	MUE ΔH_{vap} (kcal mol ⁻¹)
Parsley	0.038 ^{0.048} _{0.028}	1.18 ^{1.53} _{0.87}
Model 0	0.033 ^{0.043} _{0.024}	1.53 ^{1.87} _{1.17}
Model 1a	0.041 ^{0.052} _{0.030}	1.33 ^{1.58} _{1.11}
Model 5b	0.036 ^{0.046} _{0.027}	0.69 ^{0.86} _{0.51}
Model 5d	0.031 ^{0.041} _{0.022}	1.02 ^{1.33} _{0.73}



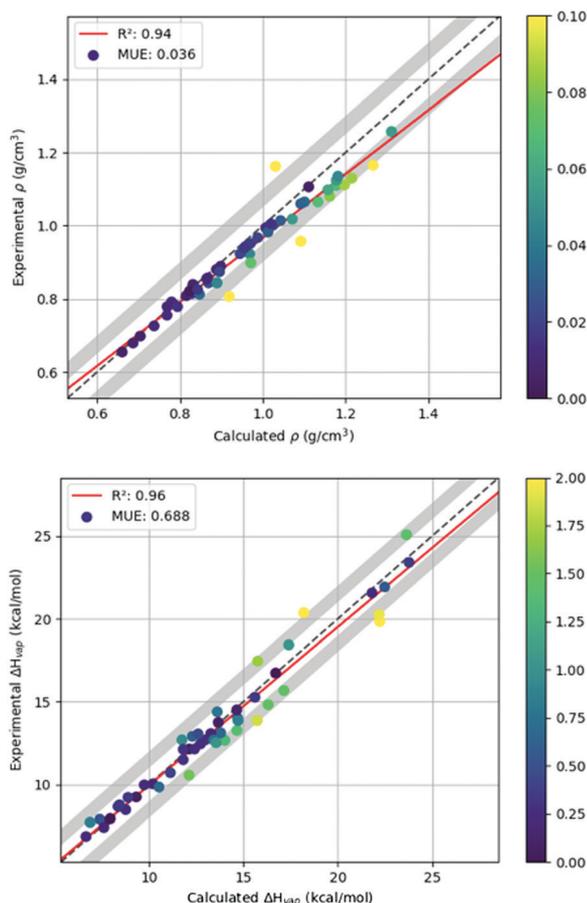


Fig. 3 Correlation between experimental and calculated (model **5b**) physical properties of liquids in the test set.

both force fields are in excellent agreement with each other and experiment ($\Delta H_{\text{vap}}^{\text{exper}} = 8.8$, $\Delta H_{\text{vap}}^{\text{QUBE}} = 8.4$ and $\Delta H_{\text{vap}}^{\text{parsley}} = 8.7$ kcal mol⁻¹). There are some small differences in charges on the terminal methyl hydrogen atoms, which add up to a large difference in charge on the carbon atoms, but this does not seem to significantly affect the liquid properties. Fig. 4(c) shows methyl isocyanate, for which QUBE model **5b** is in much better

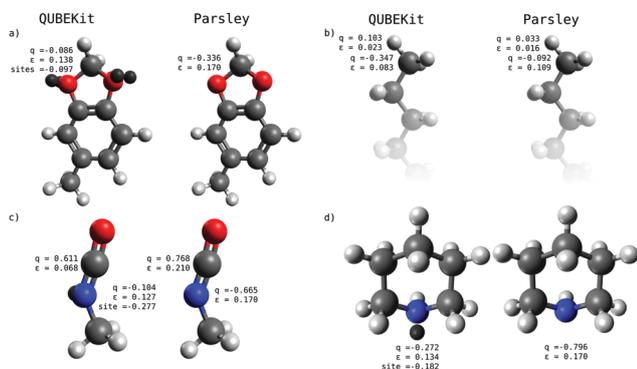


Fig. 4 A comparison of selected force field parameters between QUBE (model **5b**), and Parsley. Charges on the atom centre and virtual sites (if applicable) are shown, along with the Lennard-Jones ϵ parameter (kcal mol⁻¹).

agreement with experiment than Parsley ($\Delta H_{\text{vap}}^{\text{exper}} = 6.9$, $\Delta H_{\text{vap}}^{\text{QUBE}} = 6.7$ and $\Delta H_{\text{vap}}^{\text{parsley}} = 11.0$ kcal mol⁻¹). This seems to be a general issue with the atom-centred fixed charge force fields, since using our model **0**, we obtain $\Delta H_{\text{vap}}^{\text{QUBE}} = 10.1$ kcal mol⁻¹. Indeed, QUBE model **5b** has a smaller net charge on the N atom of methyl isocyanate than Parsley, but some of the charge is spread onto a virtual site. Finally, Fig. 4(d) shows piperidine, for which slightly improved agreement with experiment is again obtained ($\Delta H_{\text{vap}}^{\text{exper}} = 9.2$, $\Delta H_{\text{vap}}^{\text{QUBE}} = 8.9$ and $\Delta H_{\text{vap}}^{\text{parsley}} = 10.5$ kcal mol⁻¹) with a smaller net charge and a virtual site on the N atom.

We can compare the accuracy of the protocols developed here with other force fields tested on the same or similar test sets. Using the RESP2 charge assignment method, which is based on fitting to the QM electrostatic potential in implicit solvent, with optimised Lennard-Jones parameters, the density and heat of vaporisation errors on an identical test set are 0.024 g cm⁻³ and 1.67 kcal mol⁻¹.²⁷ Thus, substantial improvement in energetics is obtained in the current study, at the expense of a slight degradation in liquid density prediction. In that study, only a subset of ten Lennard-Jones parameters (the well depth and radius for each of C, O, N, H and polar H) were optimised for efficiency, though this is still more parameters than our most complex protocol (model **5b** has seven adjustable parameters). Kantonen *et al.* have used a similar QM-to-MM parameter mapping approach. They derive Lennard-Jones parameters from the MBIS partitioned atomic electron densities, with two fitting parameters (used to map the electron density decay constants onto the Lennard-Jones σ and ϵ) for each of C, O, N, H, and polar H. These QM-derived Lennard-Jones parameters are used alongside the AM1-BCC charge model, and GAFF bonded parameters. On a test set of 23 small organic molecules, they report MUEs of 0.027 g cm⁻³ and 1.1 kcal mol⁻¹ in density and heat of vaporisation, respectively, which is similar in performance to our model **5d** protocol. Visscher and Geerke have used an AIM-based QM-to-MM mapping scheme, using both C_6 and C_8 dispersion coefficients, in combination with an ESP-based charge scheme and a charge-on-spring polarisable model.³⁹ Our strategy of fitting a small number of R_i^{free} parameters was also employed there, but no off-site charges were used. Encouraging root-mean-square deviations from experiment of 0.024 g cm⁻³ and 0.39 kcal mol⁻¹ in ρ and ΔH_{vap} were reported on a combined train/test set of 49 small organic molecules.³⁹ With our automated methods for force field derivation and training, it appears that further accuracy gains can be expected by moving in future to polarisable and beyond-Lennard-Jones models.

4 Conclusions

Force field design is a lengthy process, often requiring large teams of researchers working over a period of many years. Design decisions are typically made early in the process, and any inaccuracies stemming from these assumptions propagate through to the final force field. Even the minimal Parsley force field has a set of 35 Lennard-Jones parameters,¹³ and so large training sets are required to cover sufficient chemical space.



Repeatedly performing the full training process to investigate alternative design decisions would substantially slow down the time to science. Furthermore, as we inevitably move to more complex and accurate functional forms, the size of the parameter space and danger of over-fitting will only increase.

To solve this problem, we and others have advocated making use of QM-to-MM mapping procedures to significantly reduce the number of parameters that require fitting to experimental physical properties, while retaining a small amount of empiricism to ensure accurate condensed phase properties. Though there are isolated examples in the literature, QM-to-MM mapping potentials are not widely available for general use in deriving full MM force fields for organic molecules. In QUBEKit, and its interface with ForceBalance, we provide here a means to train, test and apply these force fields, deriving parameters for the majority of terms in the force field from QM. By using AIM-based charges, Lennard-Jones parameters and virtual sites, we have a consistent set of non-bonded parameters derived from a single set of atomic electron densities, available through widely-used quantum chemistry software packages. By coupling these parameter sets through QUBEKit, we can derive bonded parameters that are specific to the molecule under study, and fully consistent with the non-bonded parameter set. We estimate that for the 51 molecules in the test set, a total of around 20k parameters are derived from QM, while only seven parameters are fit to experiment (in model 5b). Despite this minimal fitting approach, the model achieves highly competitive accuracy on a challenging test set of liquid property data.

We have used this design and train cycle to answer a number of fundamental questions about force field design protocols. It was shown that the details of the underlying QM method and implicit solvent model have negligible effect on the accuracy of condensed phase modelling with the designed force fields. That is not to say that any set of charges gives identical results, but rather that as long as the charges and Lennard-Jones parameters are co-optimised, the overall accuracy of the force field is insensitive to the particular choice of model. In this case, future efforts should focus on the cheapest QM methods that continue to give reliable results. It is well known that there is no unique method to partition the total molecular electron density into atom-centred basins. We have therefore investigated two variants of the DDEC AIM method, and the MBIS approach. Surprisingly, the older DDEC3 method showed improvements over DDEC6, but given the improvements in the latter in robustness and transferability of the charges, we continue to use this approach. The MBIS method provides a useful alternative AIM approach, available through the open source PSI4 software package. Once atomic electron densities are assigned, the details of the Lennard-Jones parameterisation also makes a difference to force field accuracy. Based on our data, we advocate setting non-zero Lennard-Jones parameters on polar hydrogen atoms, and re-scaling the strength of the QM-derived dispersion parameter. The latter approach gives not only an improvement in accuracy, but brings QM-derived and Parsley empirical Lennard-Jones parameters into closer agreement. One aspect of the Lennard-Jones interaction that we

were not able to investigate is the choice of combination rules for unlike atoms. The standard OpenMM nonbonded force, and its interface with ForceBalance, is currently limited to Lorentz–Berthelot combination rules, but it would be interesting to test alternatives in future. Finally, consistent improvement in both training and test set accuracy is achieved by introducing a small number of off-site charges to model anisotropy in the QM electron density. It is encouraging that improving the underlying physics of the force field model, with minimal costs to run time, results in such improvements.

In the current study, we have focused on liquid density and heat of vaporisation as measures of force field accuracy. However, it is important to also test force fields outside the training regime, on properties such as dielectric permittivity, binary density and enthalpy of mixing.⁷⁸ In future, we will interface the outputs of QUBEKit with Open Force Field Evaluator,⁵⁰ which will provide a more automated and efficient framework for force field testing and extend the confidence in the range of application of the force fields. More complex properties, such as free energies of hydration or even protein-ligand binding could be included (at least for testing), but would require infrastructure improvements to allow for virtual sites and the testing of a compatible water model. The current study gives confidence that this investment of time would be worthwhile. A notable feature of the current study is the number of force field models that can be rapidly designed. We have generated here 15 different models, with varying assumptions, which can be contrasted with the handful of general purpose small molecule force fields that are otherwise available. This opens up the possibility of new research into consensus property predictions using force field ensembles, which has been shown to be advantageous, for example, in recent protein-ligand binding free energy studies.⁷⁹

More broadly, we envisage QM-to-MM mapping potentials, such as these, providing synergy with traditional force fields. For example, our conclusions concerning force field design protocols can be fed into large-scale fitting efforts, such as the Open Force Field Initiative, to focus efforts in areas where accuracy improvements are expected. This will be particularly important as the community moves towards more advanced functional forms, for which the number of parameters to be fit will only increase. With the advent of machine learning models for parameterising force fields,⁸⁰ the force fields developed here could be used to provide regularisation of the parameter fits, to avoid unphysical predictions by the models. Or conversely, machine learning models could be trained to reproduce the outputs of our QM-to-MM mapping procedures to substantially reduce the computational cost of the parameterisation stage.⁸¹

All software required to derive and use the designed force fields is freely available at <https://github.com/qubekit/QUBEKit>.

Author contributions

Chris Ringrose: conceptualisation, data curation, formal analysis, investigation, methodology, software, validation, visualisation, writing – original draft. Joshua Horton: conceptualisation, data



curation, investigation, methodology, software, validation, visualisation, writing – review & editing. Lee-Ping Wang: conceptualisation, software, writing – review & editing. Daniel Cole: conceptualisation, funding acquisition, methodology, project administration, resources, supervision, writing – original draft, writing – review & editing.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

C. R. is grateful for International Partnership Seed Funding from Newcastle University. D. J. C. and J. H. acknowledge support from a UKRI Future Leaders Fellowship (grant MR/T019654/1). We thank the Open Force Field Initiative and PSI4 developers for scientific software support. This work made use of the facilities of the N8 Centre of Excellence in Computationally Intensive Research (N8 CIR) provided and funded by the N8 research partnership and EPSRC (grant EP/T022167/1).

Notes and references

- P. Dauber-Osguthorpe and A. T. Hagler, *J. Comput.-Aided Mol. Des.*, 2019, **33**, 133–203.
- A. T. Hagler, *J. Comput.-Aided Mol. Des.*, 2019, **33**, 205–264.
- J. A. Dawson, P. Canepa, T. Famprakis, C. Masquelier and M. S. Islam, *J. Am. Chem. Soc.*, 2017, **140**, 362–368.
- A. A. Franco, A. Rucci, D. Brandell, C. Frayret, M. Gaberscek, P. Jankowski and P. Johansson, *Chem. Rev.*, 2019, **119**, 4569–4627.
- L. Yang, J. T. Horton, M. C. Payne, T. J. Penfold and D. J. Cole, *J. Chem. Theory Comput.*, 2021, **17**, 5021–5033.
- Y. Olivier, J.-C. Sancho-Garcia, L. Muccioli, G. D'Avino and D. Beljonne, *J. Phys. Chem. Lett.*, 2018, **9**, 6149–6163.
- J. Nyman, O. S. Pundyke and G. M. Day, *Phys. Chem. Chem. Phys.*, 2016, **18**, 15828–15837.
- D. J. Huggins, P. C. Biggin, M. A. Dämgen, J. W. Essex, S. A. Harris, R. H. Henchman, S. Khalid, A. Kuzmanic, C. A. Loughton, J. Michel, A. J. Mulholland, E. Rosta, M. S. P. Sansom and M. W. van der Kamp, *Wiley Interdiscip. Rev.: Comput. Mol. Sci.*, 2018, **9**, e1393.
- A. S. Mey, B. K. Allen, H. E. B. Macdonald, J. D. Chodera, D. F. Hahn, M. Kuhn, J. Michel, D. L. Mobley, L. N. Naden, S. Prasad, A. Rizzi, J. Scheen, M. R. Shirts, G. Tresadern and H. Xu, *Living J. Comput. Mol. Sci.*, 2020, **2**, 18378.
- J. Wang, R. M. Wolf, J. W. Caldwell, P. A. Kollman and D. A. Case, *J. Comput. Chem.*, 2004, **25**, 1157–1174.
- K. Vanommeslaeghe, E. Hatcher, C. Acharya, S. Kundu, S. Zhong, J. Shim, E. Darian, O. Guvench, P. Lopes, I. Vorobyov and A. MacKerell Jr., *J. Comput. Chem.*, 2010, **31**, 671–690.
- W. L. Jorgensen, D. S. Maxwell and J. Tirado-Rives, *J. Am. Chem. Soc.*, 1996, **118**, 11225–11236.
- Y. Qiu, D. G. A. Smith, S. Boothroyd, H. Jang, D. F. Hahn, J. Wagner, C. C. Bannan, T. Gokey, V. T. Lim, C. D. Stern, A. Rizzi, B. Tjanaka, G. Tresadern, X. Lucas, M. R. Shirts, M. K. Gilson, J. D. Chodera, C. I. Bayly, D. L. Mobley and L.-P. Wang, *J. Chem. Theory Comput.*, 2021, **17**, 6262–6280.
- E. Boulanger, L. Huang, C. Rupakheti, A. D. MacKerell and B. Roux, *J. Chem. Theory Comput.*, 2018, **14**, 3121–3131.
- J. Yin, N. M. Henriksen, D. R. Slochower, M. R. Shirts, M. W. Chiu, D. L. Mobley and M. K. Gilson, *J. Comput.-Aided Mol. Des.*, 2016, **31**, 1–19.
- P. Xu, E. B. Guidez, C. Bertoni and M. S. Gordon, *J. Chem. Phys.*, 2018, **148**, 090901.
- O. T. Unke, S. Chmiela, H. E. Saucedo, M. Gastegger, I. Poltavsky, K. T. Schütt, A. Tkatchenko and K.-R. Müller, *Chem. Rev.*, 2021, **121**, 10142–10186.
- V. Barone, I. Cacelli, N. D. Mitri, D. Licari, S. Monti and G. Prampolini, *Phys. Chem. Chem. Phys.*, 2013, **15**, 3736.
- J. R. Maple, M.-J. Hwang, T. P. Stockfisch, U. Dinur, M. Waldman, C. S. Ewig and A. T. Hagler, *J. Comput. Chem.*, 1994, **15**, 162–182.
- T. A. Halgren, *J. Comput. Chem.*, 1996, **17**, 490–519.
- S. Sami, M. F. Menger, S. Faraji, R. Broer and R. W. A. Havenith, *J. Chem. Theory Comput.*, 2021, **17**, 4946–4960.
- J. Morado, P. N. Mortenson, M. L. Verdonk, R. A. Ward, J. W. Essex and C.-K. Skylaris, *J. Chem. Inf. Model.*, 2021, **61**, 2026–2047.
- B. Seo, Z.-Y. Lin, Q. Zhao, M. A. Webb and B. M. Savoie, *J. Chem. Inf. Model.*, 2021, **61**, 5013–5027.
- A. Jakalian, D. B. Jack and C. I. Bayly, *J. Comput. Chem.*, 2002, **23**, 1623–1641.
- L. S. Dodda, J. Z. Vilseck, J. Tirado-Rives and W. L. Jorgensen, *J. Phys. Chem. B*, 2017, **121**, 3864–3870.
- C. I. Bayly, P. Cieplak, W. Cornell and P. A. Kollman, *J. Phys. Chem.*, 1993, **97**, 10269–10280.
- M. Schauerl, P. S. Nerenberg, H. Jang, L.-P. Wang, C. I. Bayly, D. L. Mobley and M. K. Gilson, *Commun. Chem.*, 2020, **3**, 1–11.
- T. Verstraelen, S. Vandenbrande, F. Heidar-Zadeh, L. Vanduyfhuys, V. Van Speybroeck, M. Waroquier and P. W. Ayers, *J. Chem. Theory Comput.*, 2016, **12**, 3894–3912.
- T. A. Manz and N. G. Limas, *RSC Adv.*, 2016, **6**, 47771–47801.
- N. G. Limas and T. A. Manz, *RSC Adv.*, 2016, **6**, 45727–45747.
- M. Riquelme, A. Lara, D. L. Mobley, T. Verstraelen, A. R. Matamala and E. Vöhringer-Martinez, *J. Chem. Inf. Model.*, 2018, **58**, 1779–1797.
- A. Tkatchenko and M. Scheffler, *Phys. Rev. Lett.*, 2009, **102**, 6–9.
- D. J. Cole, J. Z. Vilseck, J. Tirado-Rives, M. C. Payne and W. L. Jorgensen, *J. Chem. Theory Comput.*, 2016, **12**, 2312–2323.
- J. T. Horton, A. E. Allen, L. S. Dodda and D. J. Cole, *J. Chem. Inf. Model.*, 2019, **59**, 1366–1381.
- M. Mohebifar, E. R. Johnson and C. N. Rowley, *J. Chem. Theory Comput.*, 2017, **13**, 6146–6157.
- T. A. Manz and T. Chen, *RSC Adv.*, 2019, **9**, 33310–33336.



- 37 T. Chen and T. A. Manz, *RSC Adv.*, 2019, **9**, 36492–36507.
- 38 K. M. Visscher and D. P. Geerke, *J. Chem. Theory Comput.*, 2019, **15**, 1875–1883.
- 39 K. M. Visscher and D. P. Geerke, *J. Phys. Chem. B*, 2020, **124**, 1628–1636.
- 40 S. M. Kantonen, H. S. Muddana, M. Schauerperl, N. M. Henriksen, L. P. Wang and M. K. Gilson, *J. Chem. Theory Comput.*, 2020, **16**, 1115–1127.
- 41 L.-P. Wang, J. Chen and T. Van Voorhis, *J. Chem. Theory Comput.*, 2013, **9**, 452–460.
- 42 A. E. Allen, M. C. Payne and D. J. Cole, *J. Chem. Theory Comput.*, 2018, **14**, 274–281.
- 43 T. A. Manz and D. S. Sholl, *J. Chem. Theory Comput.*, 2012, **8**, 2844–2867.
- 44 L. P. Lee, N. G. Limas, D. J. Cole, M. C. Payne, C. K. Skylaris and T. A. Manz, *J. Chem. Theory Comput.*, 2014, **10**, 5377–5390.
- 45 D. J. Cole, I. Cabeza De Vaca and W. L. Jorgensen, *Med-ChemComm*, 2019, **10**, 1116–1120.
- 46 J. T. Horton, A. E. Allen and D. J. Cole, *Chem. Commun.*, 2020, **56**, 932–935.
- 47 L. Nelson, S. Bariami, C. Ringrose, J. T. Horton, V. Kurdekar, A. S. J. S. Mey, J. Michel and D. J. Cole, *J. Chem. Inf. Model.*, 2021, **61**, 2124–2130.
- 48 T. Fröhlking, M. Bernetti, N. Calonaci and G. Bussi, *J. Chem. Phys.*, 2020, **152**, 230902.
- 49 C. J. Fennell, K. L. Wymer and D. L. Mobley, *J. Phys. Chem. B*, 2014, **118**, 6438–6446.
- 50 S. Boothroyd, L.-P. Wang, D. L. Mobley, J. D. Chodera and M. R. Shirts, *J. Chem. Theory Comput.*, 2022, **18**, 3566–3576.
- 51 M. P. Oliveira, M. Andrey, S. R. Rieder, L. Kern, D. F. Hahn, S. Riniker, B. A. C. Horta and P. H. Hünenberger, *J. Chem. Theory Comput.*, 2020, **16**, 7525–7555.
- 52 M. D. Pierro, M. L. Mugnai and R. Elber, *J. Phys. Chem. B*, 2014, **119**, 836–849.
- 53 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, B. Mennucci, G. A. Petersson, H. Nakatsuji, M. Caricato, X. Li, H. P. Hratchian, A. F. Izmaylov, J. Bloino, G. Zheng, J. L. Sonnenberg, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, J. A. Montgomery, J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd, E. Brothers, K. N. Kudin, V. N. Staroverov, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, N. Rega, J. M. Millam, M. Klene, J. E. Knox, J. B. Cross, V. Bakken, C. Adamo, J. Jaramillo, R. Gomperts, R. E. Stratmann, O. Yazyev, A. J. Austin, R. Cammi, C. Pomelli, J. W. Ochterski, R. L. Martin, K. Morokuma, V. G. Zakrzewski, G. A. Voth, P. Salvador, J. J. Dannenberg, S. Dapprich, A. D. Daniels, Ö. Farkas, J. B. Foresman, J. V. Ortiz, J. Cioslowski and D. J. Fox, *Gaussian 09 Revision A.2*, 2009.
- 54 J. Wang, W. Wang, P. A. Kollman and D. A. Case, *J. Mol. Graphics Modell.*, 2006, **25**, 247–260.
- 55 S. Riniker and G. A. Landrum, *J. Chem. Inf. Model.*, 2015, **55**, 2562–2574.
- 56 G. Landrum, RDKit: Open-source cheminformatics, <http://www.rdkit.org/>.
- 57 X. Gao, F. Ramezanghorbani, O. Isayev, J. S. Smith and A. E. Roitberg, *J. Chem. Inf. Model.*, 2020, **60**, 3408–3415.
- 58 C. Bannwarth, E. Caldeweyher, S. Ehlert, A. Hansen, P. Pracht, J. Seibert, S. Spicher and S. Grimme, *Wiley Interdiscip. Rev.: Comput. Mol. Sci.*, 2020, **11**, e1493.
- 59 P. Eastman, J. Swails, J. D. Chodera, R. T. McGibbon, Y. Zhao, K. A. Beauchamp, L.-P. Wang, A. C. Simmonett, M. P. Harrigan, C. D. Stern, R. P. Wiewiora, B. R. Brooks and V. S. Pande, *PLoS Comput. Biol.*, 2017, **13**, e1005659.
- 60 D. G. A. Smith, L. A. Burns, A. C. Simmonett, R. M. Parrish, M. C. Schieber, R. Galvelis, P. Kraus, H. Kruse, R. Di Remigio, A. Alenaizan, A. M. James, S. Lehtola, J. P. Misiewicz, M. Scheurer, R. A. Shaw, J. B. Schriber, Y. Xie, Z. L. Glick, D. A. Sirianni, J. S. O'Brien, J. M. Waldrop, A. Kumar, E. G. Hohenstein, B. P. Pritchard, B. R. Brooks, H. F. Schaefer, A. Y. Sokolov, K. Patkowski, A. E. DePrince, U. Bozkaya, R. A. King, F. A. Evangelista, J. M. Turney, T. D. Crawford and C. D. Sherrill, *J. Chem. Phys.*, 2020, **152**, 184108.
- 61 D. G. Smith, A. T. Lolinco, Z. L. Glick, J. Lee, A. Alenaizan, T. A. Barnes, C. H. Borca, R. Di Remigio, D. L. Dotson and S. Ehlert, *et al.*, *J. Chem. Phys.*, 2021, **155**, 204801.
- 62 D. L. Mobley, É. Dumont, J. D. Chodera and K. A. Dill, *J. Phys. Chem. B*, 2007, **111**, 2242–2254.
- 63 J. C. A. Prentice, J. Aarons, J. C. Womack, A. E. A. Allen, L. Andrinopoulos, L. Anton, R. A. Bell, A. Bhandari, G. A. Bramley, R. J. Charlton, R. J. Clements, D. J. Cole, G. Constantinescu, F. Corsetti, S. M.-M. Dubois, K. K. B. Duff, J. M. Escartín, A. Greco, Q. Hill, L. P. Lee, E. Linscott, D. D. O'Regan, M. J. S. Phipps, L. E. Ratcliff, Á. R. Serrano, E. W. Tait, G. Teobaldi, V. Vitale, N. Yeung, T. J. Zuehlsdorff, J. Dziedzic, P. D. Haynes, N. D. M. Hine, A. A. Mostofi, M. C. Payne and C.-K. Skylaris, *J. Chem. Phys.*, 2020, **152**, 174111.
- 64 J. C. Womack, L. Anton, J. Dziedzic, P. J. Hasnip, M. I. J. Probert and C.-K. Skylaris, *J. Chem. Theory Comput.*, 2018, **14**, 1412–1432.
- 65 A. D. MacKerell, D. Bashford, M. Bellott, R. L. Dunbrack, J. D. Evanseck, M. J. Field, S. Fischer, J. Gao, H. Guo, S. Ha, D. Joseph-McCarthy, L. Kuchnir, K. Kuczera, F. T. K. Lau, C. Mattos, S. Michnick, T. Ngo, D. T. Nguyen, B. Prodhom, W. E. Reiher, B. Roux, M. Schlenkrich, J. C. Smith, R. Stote, J. Straub, M. Watanabe, J. Wiórkiewicz-Kuczera, D. Yin and M. Karplus, *J. Phys. Chem. B*, 1998, **102**, 3586–3616.
- 66 A. Nikitin, *J. Comput.-Aided Mol. Des.*, 2019, **34**, 437–441.
- 67 W. L. Jorgensen, J. Chandrasekhar, J. D. Madura, R. W. Impey and M. L. Klein, *J. Chem. Phys.*, 1983, **79**, 926–935.
- 68 X. C. Yan, M. J. Robertson, J. Tirado-Rives and W. L. Jorgensen, *J. Phys. Chem. B*, 2017, **121**, 6626–6636.
- 69 I. S. Gutiérrez, F.-Y. Lin, K. Vanommeslaeghe, J. A. Lemkul, K. A. Armacost, C. L. Brooks and A. D. MacKerell, *Bioorg. Med. Chem.*, 2016, **24**, 4812–4825.
- 70 M. H. Kolář and P. Hobza, *Chem. Rev.*, 2016, **116**, 5155–5187.
- 71 Y. Qiu, D. G. Smith, C. D. Stern, M. Feng, H. Jang and L. P. Wang, *J. Chem. Phys.*, 2020, **152**, 244116.



- 72 A. E. Allen, M. J. Robertson, M. C. Payne and D. J. Cole, *ACS Omega*, 2019, **4**, 14537–14550.
- 73 M. J. Robertson, J. Tirado-Rives and W. L. Jorgensen, *J. Chem. Theory Comput.*, 2015, **11**, 3499–3509.
- 74 J. Řezáč, D. Bím, O. Gutten and L. Rulíšek, *J. Chem. Theory Comput.*, 2018, **14**, 1254–1266.
- 75 J. B. Foresman, T. A. Keith, K. B. Wiberg, J. Snoonian and M. J. Frisch, *J. Phys. Chem.*, 1996, **100**, 16098–16104.
- 76 J. Tomasi, B. Mennucci and R. Cammi, *Chem. Rev.*, 2005, **105**, 2999–3094.
- 77 D. L. Mobley, C. C. Bannan, A. Rizzi, C. I. Bayly, J. D. Chodera, V. T. Lim, N. M. Lim, K. A. Beauchamp, D. R. Slochower, M. R. Shirts, M. K. Gilson and P. K. Eastman, *J. Chem. Theory Comput.*, 2018, **14**, 6076–6092.
- 78 S. Boothroyd, O. Madin, D. Mobley, L.-P. Wang, J. Chodera and M. Shirts, *J. Chem. Theory Comput.*, 2022, **18**, 3577–3592.
- 79 V. Gapsys, L. Pérez-Benito, M. Aldeghi, D. Seeliger, H. van Vlijmen, G. Tresadern and B. L. de Groot, *Chem. Sci.*, 2020, **11**, 1140–1152.
- 80 Y. Wang, J. Fass and J. D. Chodera, 2020, arXiv preprint arXiv:2010.01196.
- 81 P. Bleiziffer, K. Schaller and S. Riniker, *J. Chem. Inf. Model.*, 2018, **58**, 579–590.

