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# Insight into the role of reduced graphene oxide in enhancing photocatalytic hydrogen evolution in disordered carbon nitride†

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Compared to crystalline carbon nitride, the performance of disordered carbon nitride (d-CN) as a hydrogen production photocatalyst is extremely poor. Owing to its disordered atomic orientation, it is prone to numerous defect states. These energy states are potential sites for trapping and recombination of photogenerated charge carriers. As a result, rapid recombination of photogenerated charge carriers places a fundamental photophysical challenge in charge separation and transport, which inhibits its photocatalytic activity. In the presence of reduced graphene oxide (rGO), d-CN shows enhanced photocatalytic production of hydrogen. However, photophysical insight into the tacit role of rGO is not well understood which limits the rational design of d-CN as a photocatalyst. Particularly, understanding of the early time-scale (in fs to ps) recombination mechanism and the charge transport kinetics has not yet been achieved. With the help of ultrafast transient absorption spectroscopy, femtosecond time-resolved photoluminescence spectroscopy and transient photocurrent measurements, this article deciphers the ultrafast dynamics of the separation and transport of photogenerated charge carriers in d-CN facilitated by rGO. It is found that rGO substantially suppresses the bimolecular and trap-assisted recombination and enables a faster separation of charge carriers. As a result, it increases the lifetime of the charge carriers to be transported to the surface catalytic sites, and therefore, augments the rate of hydrogen production almost by an order of magnitude. Our findings therefore offer a proof-of-concept for overcoming the trap-mediated recombination problems in disordered carbon nitride.

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## 1. Introduction

Carbon nitride, arguably, is the most popular metal-free polymeric semiconductor for various energy related conversion and storage applications, particularly for photocatalytic hydrogen production from water.<sup>1,2</sup> Broadly, carbon nitride as a photocatalyst can be categorized into two types, namely, crystalline and amorphous carbon nitride. Due to its disordered atomic orientation and resultant defect states, amorphous carbon nitride is also termed as disordered carbon nitride (d-CN). Crystalline carbon nitride is an intensively studied photocatalyst evident from hundreds of archival publications, while there are only a few publications (<10 articles) on amorphous carbon nitride or d-CN.<sup>1,2</sup> Considering the apparent quantum efficiency (AQE) of hydrogen production, d-CN is extremely

poor (AQE < 6%) compared with crystalline carbon nitride (AQE > 60%).<sup>3,4</sup>

The poor photocatalytic performance of d-CN is mainly attributed to its poor transport of charge carriers.<sup>5-7</sup> This is largely due to the non-aromatic connection between the polymer strands of CN which causes photogenerated singlet excitons to be typically confined to the tri-s-triazine (or heptazine) unit of CN.<sup>8</sup> The excitons in CN therefore experience a larger binding energy ( $\gg k_B T$ , where  $k_B$  is the Boltzmann constant and  $T$  is the temperature in Kelvin; at room temperature,  $k_B T$  is  $\sim 0.26$  eV) to be split into free electrons and holes due to electron-lattice and electron-electron Coulomb interactions.<sup>8,9</sup> To achieve a charge separated state, the excitons should be dissociated within their lifetime, which is in the picosecond time scale.<sup>10,11</sup> Additionally, the carriers must survive recombination before reaching the catalytic sites.<sup>7</sup>

The charge transport becomes further exacerbated by the residual chemical defects (e.g., dangling bonds, interstitials, vacancies, etc.) in the crystal structure due to the incomplete polycondensation of precursor monomers or doping/adsorption of heteroatoms in C-N networks.<sup>4,12</sup> Pendant amine groups are the

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root cause of surface dangling bonds that create shallow trap states in CN and act as potential sites for charge carrier trapping.<sup>3,13</sup> Being in this regime, charge trapping detrimentally reduces the charge carrier mobility and makes the interfacial charge transfer difficult. It causes a kinetic competition between charge separation and non-radiative recombination at the semiconductor–water interfaces, and therefore, presents a significant photophysical challenge.<sup>14</sup>

The efficiency of charge extraction depends on the ratio of the rate constants for charge recombination and charge separation.<sup>5</sup> Generally, a high rate of hydrogen production requires charge recombination processes to occur on a longer time scale than the charge separation and extraction processes.<sup>15,16</sup> It is therefore important to determine the time scale for charge recombination processes. Transient absorption spectroscopy (TAS) and time-resolved photoluminescence (TRPL) spectroscopy are the widely accepted techniques used to assess the individual dynamics of interfacial charge transfer processes.<sup>5,17</sup>

Because pristine d-CN is a poor photocatalyst,<sup>18</sup> most often a noble metal co-catalyst (*e.g.* Pt) is employed to enhance the rate of hydrogen production on the surface of d-CN.<sup>19</sup> Given the scarcity and price of noble metals, metal-free co-catalysts are highly desirable. In lieu of Pt, incorporation of reduced graphene oxide (rGO) has been shown to enhance the performance of the catalyst.<sup>20–24</sup> However, little is known about how rGO contributes to the overall charge transport process. It is therefore at the research forefront to understand what limits the performance of d-CN and why the incorporation of rGO enhances its performance.

This situation demands a systematic research endeavor to probe into the kinetics of photoexcited charge carriers and to delineate the directional transfer of the photoexcited electrons between the photocatalyst and co-catalyst or *vice versa*. In this regard, here, we employed femtosecond TAS and TRPL spectroscopy complemented by transient photocurrent measurements to provide insight into the charge separation and transport of d-CN facilitated by rGO for photocatalytic hydrogen production.

## 2. Results and discussion

### 2.1 Synergistic physicochemical effects is induced by the overlapped rGO nanolayers on d-CN

We synthesized d-CN *via* polycondensation of thiourea and dicyandiamide at 620 °C using slightly modified synthesis procedures reported previously.<sup>3</sup> The d-CN/rGO hybrid was synthesized by mixing rGO powder (Sigma-Aldrich) with the CN precursors following the same procedures of temperature treatment (see the Experimental section in the ESI† for details). The as-synthesized d-CN shows a morphology of disorderly stacked nanosheets under a transmission electron microscope (TEM; Fig. 1A). We did not see any noticeable changes in the morphology under the TEM after incorporation of rGO into the d-CN skeleton (Fig. 1B). This is also confirmed from the identical X-ray diffractogram (Fig. S1, ESI†). However, when compared with d-CN, the XRD pattern of d-CN/rGO shows a relatively flat peak at  $\sim 13^\circ$  which describes the in-plane

structural packing motif and a broader inter-planar stacking peak at  $\sim 27.0^\circ$ . These reflections indicate that rGO is not connected to the lattice of d-CN but overlapped (also confirmed from the TEM images, Fig. 1A and B) and is connected to the surface terminal groups.<sup>25,26</sup> In support of this observation, in the FTIR spectra, the C–O stretching vibrations of epoxy groups at  $1096\text{ cm}^{-1}$  and the C=O stretching vibrations of COOH groups at  $1725\text{ cm}^{-1}$  for rGO are absent in d-CN/rGO, but the vibrations of typical C–N heterocycles (C–N, C=N) at  $1000$  to  $1500\text{ cm}^{-1}$  are present (see Fig. S2, ESI†).<sup>27</sup> This is further supported by the XPS deconvoluted C1s spectrum (Fig. S3, ESI†) which consists only of C=C–C and N=C–N groups.<sup>28</sup>

The absence of epoxy, carboxyl and carboxylic groups is indicative of the functionalization of the rGO periphery with pyridinic and pyrrolic nitrogen when N-rich precursors (thiourea and dicyandiamide) and rGO were copolymerized onto the d-CN/rGO hybrid.<sup>29</sup> We previously reported<sup>3</sup> that, during the synthesis of amorphous carbon nitride from polycondensation of N-rich precursors, significant amounts of ammonia were eliminated which could possibly destroy the regular stacking of rGO. A decrease of *d*-spacing ( $\sim 0.05\text{ nm}$ ) in the case of d-CN/rGO is observed, which infers the compact overlapped layers between rGO and d-CN (Fig. S1, ESI†). A synergistic physicochemical effect can be expected due to this strong coupling between d-CN and rGO.

### 2.2 Incorporation of rGO red-shifts the visible light absorption

UV-Vis absorption spectroscopy gives further details about the strength of the electronic transition between the valence band (VB) and the conduction band (CB). A smooth and monotonically increasing absorption with decreasing wavelength marks the indirect bandgap transitions for both d-CN and d-CN/rGO (Fig. 1C). Unlike direct bandgap transitions, the indirect bandgap transitions require phonon assistance or changes in the crystal momentum (*k*-vector) which invariably influence the relaxation of excited states due to an electron–phonon coupling effect.<sup>30</sup>

Relative to d-CN, a slightly red-shifted absorption peak in the case of d-CN/rGO infers that the electronic transitions from the VB to the excitonic states in the CB or the states located just below the bottom of the CB will require a lower exciton binding energy to be overcome.<sup>31</sup> We have extrapolated the bandgap of d-CN (2.76 eV) and d-CN/rGO (2.66 eV) from the Kubelka–Munk (*K*–*M*) plot (Fig. S4, ESI†). We and other groups have also shown that the decomposition of N-rich precursors to CN incorporates nitrogen atoms into the edges and induces defect sites in rGO.<sup>27</sup> Particularly, the graphitic-N links CN and rGO which reduces the electrical resistivity of the resulting hybrid.<sup>32</sup> This would positively affect the charge carrier photogeneration and transfer. We therefore expect that d-CN/rGO would be a functional photocatalyst.

### 2.3 rGO enhances the rate of visible light ( $\lambda \geq 420\text{ nm}$ ) H<sub>2</sub> production by d-CN

We measured the visible light (from a 300 W Xe lamp mounted with a 420 nm cut-off filter) photocatalytic H<sub>2</sub> production activities of the d-CN/rGO system in a batch reactor where 50 mg of the catalyst was dispersed in 80 mL of an aqueous



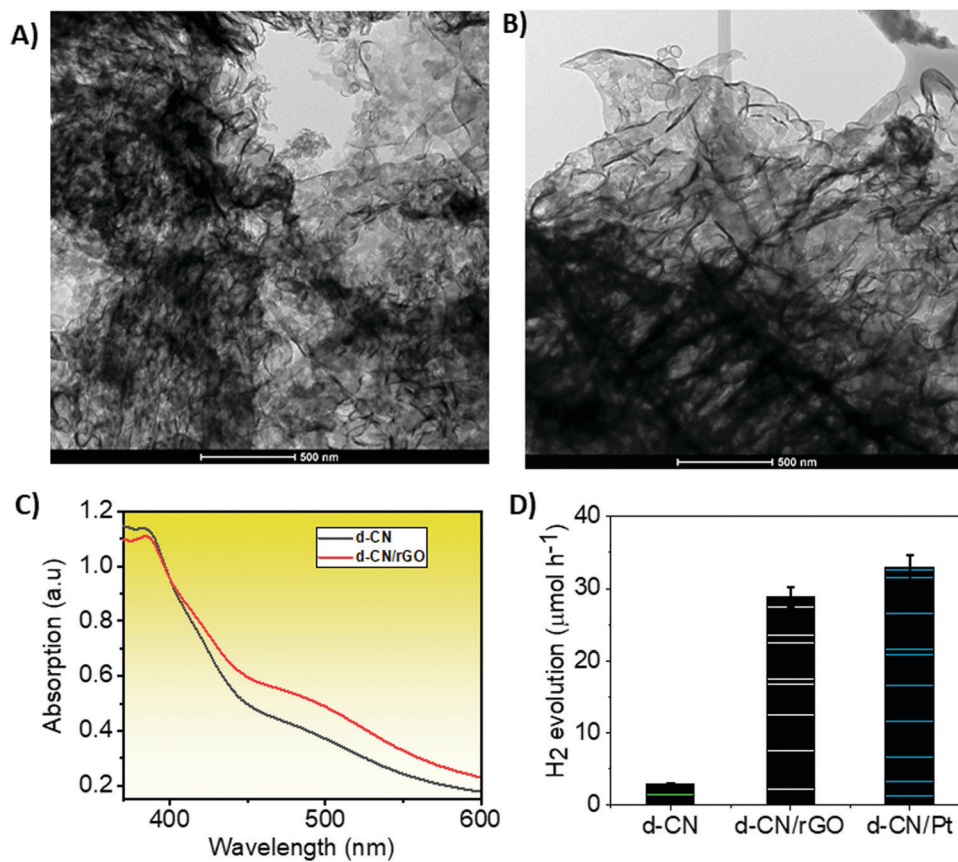


Fig. 1 Morphology under an electron microscope, and optical absorption and photocatalytic performance. TEM images of (A) d-CN and (B) d-CN/rGO. (C) Combined UV-vis absorption spectra. (D) Rate of hydrogen evolution by d-CN, d-CN/rGO, and d-CN/Pt from 10 vol% triethanolamine solution when irradiated by a 300 W Xe lamp mounted with a 420 nm cut-off filter at STP.

solution containing 10 vol% triethanolamine as a sacrificial reagent. The rate of H<sub>2</sub> production was quantified using gas chromatography. Bare d-CN showed a H<sub>2</sub> production rate of 2.9 μmol h<sup>-1</sup>, while d-CN/rGO showed 28.8 μmol h<sup>-1</sup> (Fig. 1B). We observed a gradual increase of activity with increasing the ratio of d-CN to rGO, which reaches a peak at 10 : 1. We observed a decreasing trend of activity for any further increase in the rGO content. d-CN also shows a comparable hydrogen production rate to d-CN/Pt (32.9 μmol h<sup>-1</sup>), and there was no noticeable degradation in cyclic hydrogen production for 24 h (data are not shown here, but available on request). As can be seen, there was almost an order of magnitude increase in the H<sub>2</sub> production rate in d-CN/rGO compared to bare d-CN. Intuitively, this may be due to the favorable charge transport dynamics that stems from the synergistic effect between rGO and d-CN. We therefore examined the d-CN/rGO system with steady-state and femtosecond TRPL spectroscopy, followed by TAS and transient photocurrent measurements, to understand the charge transport mechanisms.

## 2.4 Photophysical insight into the role of rGO in enhancing the photocatalytic performance of d-CN

**2.4.1 rGO passivates the trap states and reduces the radiative/trap-assisted recombination of photogenerated charge carriers in d-CN.** After examining the UV-vis absorption behavior, we

investigated the extent of photogenerated charge carrier recombination (band-edge and/or trap assisted) using steady state photoluminescence (PL). After exciting the semiconducting sample with incident light with  $h\nu > E_g$ , the photogenerated electrons in the conduction band have the tendency to fall back to the valence band to maintain equilibrium by releasing energy as heat and/or photons.<sup>33</sup> PL spectroscopy is a convenient technique to track the radiative charge carrier recombination. In principle, the PL intensity is proportional to the number of photons emitted from an excited state.<sup>31</sup> Therefore, the PL spectrum could be a superposition of band-edge emission and trap state emission. In agreement with previous reports, the acquired PL spectra of d-CN and d-CN/rGO show band edge emission near 440 nm which involves transition from  $\pi$  conjugated states and intra-bandgap state emission near 500 nm involving tertiary N atoms (Fig. 2A–C).<sup>34,35</sup>

Although the PL intensity is dominated by the band-edge emission for typical semiconductor nanoparticles, the PL spectrum can also be used as a sensitive probe to extract useful information about bandgap states. Trap states are typically located within the bandgap, and consequently photons emitted from trapped electrons should exhibit a red-shifted emission relative to band-edge emission. We deconvoluted the steady state PL spectra using a Gaussian fit in order to qualitatively



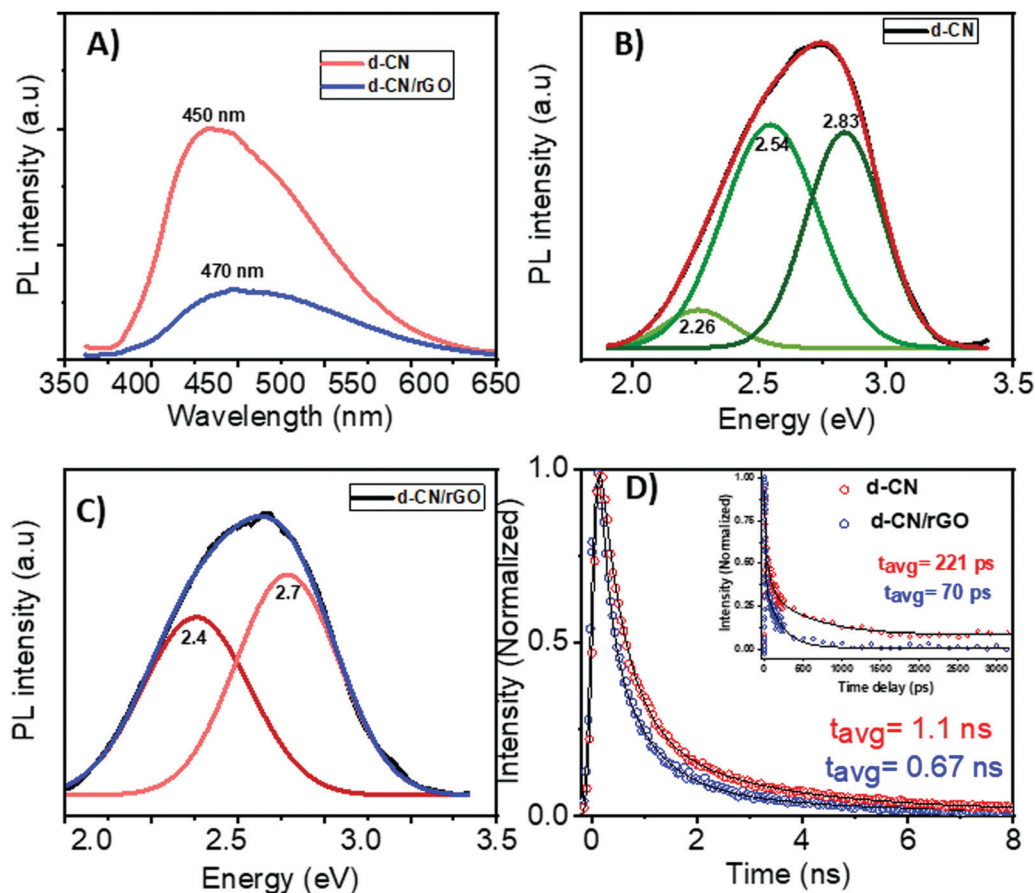


Fig. 2 Steady state and time-resolved photoluminescence (PL) measurements. (A) Combined steady state PL spectra of d-CN and d-CN/rGO. Deconvoluted steady state PL spectra of (B) d-CN and (C) d-CN/rGO. (D) TRPL decay traces for d-CN and d-CN/rGO measured in TCSPC and fluorescence upconversion (inset) in response to 370 nm optical excitation. Solid black lines represent the exponential fitting of the experimental data.

understand the contribution of band-edge emission and trap state emission. As can be seen in Fig. 2B, in line with previous studies,<sup>34,35</sup> the band centered near 2.83 eV is attributed to the bandgap emission, and those at 2.54 and 2.26 eV are attributed to intra-bandgap state (shallow and deep traps, respectively) emission for d-CN. From the K–M plot, the bandgap for d-CN was extrapolated to be 2.76 eV (Fig. S4, ESI†). Therefore, the intra-bandgap emissive states are  $\sim 0.22$  eV below the absorption band edges for shallow traps and  $\sim 0.5$  eV for deep traps.<sup>36</sup> The deconvoluted PL spectrum for d-CN/rGO is shown in Fig. 2C. For d-CN/rGO, the peak centered at 2.7 eV is assigned to bandgap emission and the peak centered at 2.4 eV is assigned to the emission from shallow trap states. Both of these emission peaks are red-shifted when compared with that of d-CN, as shown in Fig. 2B. This is also in good agreement with the corresponding absorption spectra shown in Fig. 1C. Given the K–M extrapolated bandgap of 2.66 eV for d-CN/rGO, intra-bandgap emissive states are  $\sim 0.26$  eV below the absorption band edges which may be assigned to shallow traps or emissive states associated with rGO. The absence of deep trap state emission in the case of d-CN/rGO implies that incorporation of rGO is likely to passivate the defects in d-CN (see also ESI† note S1). We have estimated the highest amount of energy

contained in the excited state to be 2.92 eV and 2.87 eV from the cross-over of the normalized UV-vis and PL spectra that cross at 424 nm and 432 nm for d-CN and d-CN/rGO, respectively (see Fig. S5, ESI†).<sup>13</sup> Therefore, excited state electrons in d-CN are seemingly more prone to undergo electron–electron and electron–lattice interactions before relaxation to the conduction band minimum.

Comparing the steady-state PL spectra of d-CN with and without rGO, we can see the drastic reduction in the intensity of the respective PL spectra. As the PL intensity is proportional to the emitted photons due to electron–hole recombination, we can say that radiative recombination of photogenerated electron–hole pairs in d-CN/rGO is significantly reduced relative to bare d-CN (see also ESI† note S1). However, steady-state PL does not provide information on the relaxation and trapping of excited state electrons before falling back to the ground state, and therefore, cannot provide decisive information on whether the trap assisted recombination dominates radiative recombination or/and charge carrier separation happens before recombination. Therefore, we used time resolved photoluminescence (TRPL) and absorption measurements in the femtosecond to nanosecond time regime to understand the trapping and trap-assisted recombination of charge carriers. Details of femtosecond



transient absorption spectroscopy (TAS) and fluorescence upconversion can be found in our previous reports<sup>37–39</sup> as well as in the ESI†

**2.4.2 rGO favors the ultrafast (fs to ps) charge transport dynamics in d-CN: kinetic competition between recombination, separation and transport of photogenerated charge carriers.** The TRPL measurements were carried out using time correlated single photon counting (TCSPC) and fluorescence upconversion techniques to understand the band-edge recombination kinetics before and after insertion of reduced graphene oxide. As can be seen in steady-state PL spectra, the PL bands centered at  $\sim 450$  and  $\sim 470$  nm correspond to the exciton band-edge emission for d-CN and d-CN/rGO, respectively. It is obvious that the loading of rGO resulted in substantial PL quenching of d-CN. This is because the incorporation of rGO aids in quenching the excited state electrons in d-CN as evidently shown by fluorescence upconversion and TAS data (see below). The time-resolved PL kinetics measured in TCSPC (excitation at 370 nm and emission at 460 nm) showed that rGO resulted in shortening of the PL lifetime by a factor of 1.7 (*cf.*  $0.67 \pm 0.03$  ns for d-CN/rGO *versus*  $1.1 \pm 0.05$  ns for bare d-CN) (Fig. 2D). Moreover, the mean PL lifetimes of  $\sim 70 \pm 3.5$  ps for d-CN/rGO and  $> 221 \pm 11$  ps for d-CN were extracted from the fluorescence upconversion decay traces (Fig. 2D inset). Together with PL quenching, the acceleration of the band-edge PL decay kinetics could be attributed to interfacial carrier transfer from d-CN to rGO.

TAS helps us to resolve the temporal transfer of electrons on an ultrafast time-scale from excited states to trap or interfacial states (see also ESI† note S2). The fs-TA measurements were performed in the presence of 10% TEOA in water. In our TAS measurements, we have chosen 370 nm as the excitation wavelength of the femtosecond pump pulses to promote the electrons effectively from the valence band to the conduction band of d-CN. Fig. 3A and B present the fs-TA spectra at different time delays of d-CN and d-CN/rGO in response to 370 nm optical excitation. In both cases, persistent negative (known as ground state bleach, GSB) and positive (known as photoinduced absorption, PIA) absorption changes can be discerned. The TA signals in the NIR region and visible region are due to photoexcited electrons and a combination of electrons and holes, respectively. The GSB signal (in the visible region) is contributed by electrons and holes. In the presence of TEOA the hole concentration is decreased; however, the electron concentration remains unchanged. That is why the signal intensity in the low wavelength region is reduced (and appears as a low S/N) as compared to the electron signal in the NIR region. There are two main reasons for the observed differences in the S/N ratio between the visible and NIR regions: (i) the probe light intensity after absorption of the sample is lower in the visible as compared to the NIR region, which is due to d-CN dominating absorption in the visible region (Fig. 1C); (ii) the absorption cross-section of a conduction band electron is greater than that of a valence band hole. Moreover, d-CN absorbs light in the visible region, which affects unequally the probe light intensity in both regions.

The GSB spectral position consistent with steady-state absorption of the  $S_1$  excitonic band which manifests the bleaching of the

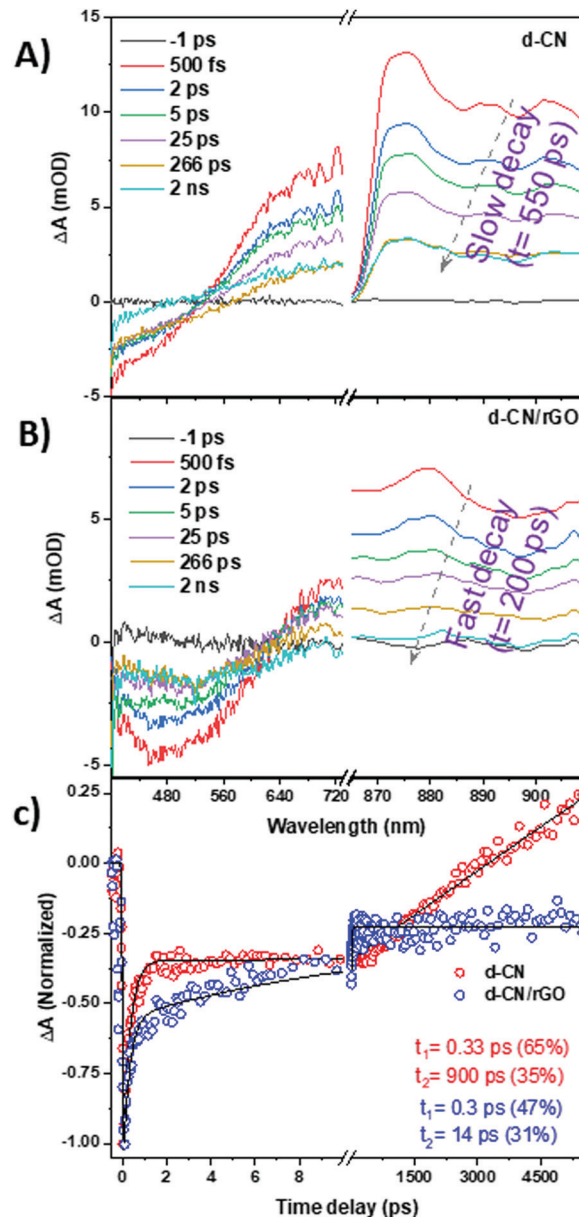


Fig. 3 Transient absorption spectroscopy. Femtosecond transient absorption spectra at different time delays for (A) d-CN and (B) d-CN/rGO in response to 370 nm excitation. (C) Normalized bleach recovery kinetics of d-CN and d-CN/rGO at 435 nm in response to 370 nm excitation. Solid black lines represent the exponential fitting of the experimental data.

ground state is due to depletion of valence band (VB) electrons and appears from state filling at the bandgap.<sup>37</sup> The broad PIA is attributed to the absorption of excited state electrons generated in the CB.<sup>40,41</sup> The excited electrons are not different in the IR region. However, in the TA spectra it appears as a positive band (known as photoinduced absorption) in the IR region and a negative band in the visible region (GSB) because of different directions of transition. The NIR spectrum in Fig. 3B shows the excited electron in the CB of d-CN not rGO. The excited state dynamics of rGO is much faster (hundreds of fs to a few ps) than



what we presented in Fig. 3. The rGO concentration is 10 wt% of d-CN, and therefore upon excitation, mostly d-CN is excited, not rGO. In the case of d-CN, as can be seen in Fig. 3A, the GSB signal gradually recovers and in 3 ns it reaches zero. On the other hand, the GSB for d-CN/rGO (Fig. 3B) shows also fast recovery at an early time scale; however, it remains unchanged for a longer time, which could be attributed to efficient charge separation in the d-CN/rGO composite. In contrast, the PIA of d-CN/rGO decays at a faster rate than d-CN. In the present study, we are interested in the charge separation dynamics in d-CN/rGO. That is why our TA measurements were focused on in the sub-ns time scale. As can be seen the excited state lifetime of d-CN is  $> 5$  ns.

To explore the charge transfer behavior, we compared the GSB and PIA kinetics in Fig. 3C and Fig. S6 (ESI<sup>†</sup>), respectively. With an exponential fitting of the kinetic traces acquired at 435 nm, we have retrieved the characteristic relaxation time constants. The exponential fitting results are  $\tau_1 = 0.33 \pm 0.01$  ps (65%) and  $\tau_2 = 900 \pm 40$  ps (35%) for bare d-CN, which gives a mean relaxation lifetime of  $\sim 315 \pm 15$  ps. The two decay times are indicative of exciton relaxation pathways to two trap states with different trap depths, namely shallow traps and deep traps.<sup>31,42</sup> Considering the proton reduction potential of  $-410$  mV vs. SHE at pH = 7 (pure water) or  $-590$  mV vs. SHE at pH = 10 (for the triethanolamine–water mixture), a loss of 0.6 eV by deeply trapped electrons would therefore result in a substantial loss of driving force for proton reduction regardless of the type of solution. It is therefore not surprising that d-CN shows a low performance in the production of hydrogen.

In contrast, the GSB kinetics can fit with two time constants  $\tau_1 = 0.30 \pm 0.01$  ps (47%) and  $\tau_2 = 14 \pm 0.6$  ps (31%) for d-CN/rGO. Similar to bare d-CN the very fast component (0.30 ps) remains unchanged, which indicates de-excitation in d-CN/rGO associated through the trapping mediated process. However, the weight percentage is reduced as compared to the bare d-CN, confirming that trap assisted processes are reduced as electrons are transferred from d-CN to the conduction band of rGO. In addition, the time constant of 14 ps represents a quick charge recombination between separated charge carriers generated after electron transfer. The remaining charge recombination ( $\sim 22\%$ ) in d-CN/rGO occurs at a  $> 5$  ns time scale, clearly representing a longer charge carrier lifetime.

Normalized TA kinetics at 870 nm of d-CN and d-CN/rGO is presented in Fig. S6 (ESI<sup>†</sup>). The kinetics of bare d-CN in the NIR region (*e.g.*, 870 nm) is due to excited electrons, which undergo trapping during the deexcitation process. The intermediate trap state lifetime is higher as compared to the initial excited state. That is why the rise is observed in PIA decay in the NIR region after a 1 ns time delay. Fig. 3C highlights the kinetics comparison of d-CN and d-CN/rGO at the GSB position (435 nm). It can be seen that the GSB kinetics of bare d-CN recovers completely and appears as a positive signal at a  $> 5$  ns time delay. In addition to Fig. S6, Fig. 3C confirms the presence of a mid-band longer lived trap state in bare d-CN.

It can be seen that the excited state of d-CN/rGO decays faster than that of bare d-CN, clearly suggesting photoexcited electron transfer, which is consistent with the decay measured from TRPL (Fig. 2D). The slower charge recombination and faster excited state decay in d-CN/rGO are attributed to the fast interfacial transfer and efficient charge separation between d-CN and rGO. The very fast electron transfer (within the temporal resolution,  $< 100$  fs) from d-CN to rGO and slow charge recombination ( $> 5$  ns) are the key factors for the enhancement of the water reduction reaction in d-CN/rGO. It can be safely inferred according to the data extracted from TA spectroscopy that incorporation of rGO into d-CN induces a channel for excited electron transfer from d-CN to rGO, and therefore, provides a means for enhancement of hydrogen production.

**2.4.3 rGO accelerates the interfacial charge transfer at the solid–liquid junction.** We have further probed the efficiency of charge separation and transport by measuring the corresponding transient photocurrent. In principle, the transient photocurrent response is caused by the build-up and decay of the minority carrier concentration reaching the surface. The transient photocurrent response could therefore gauge the competition between charge transfer and recombination.<sup>43</sup> We have observed an  $\sim 3$  fold increase in photocurrent density ( $4.3 \mu\text{A cm}^{-2}$  vs.  $12.0 \mu\text{A cm}^{-2}$ ) in the case of d-CN/rGO which clearly attests to the more efficient separation of the photogenerated charge carriers in d-CN/rGO (Fig. 4A). This manifests that rGO could serve as an electron sink<sup>44</sup> to retard the back electron transfer to d-CN, therefore suppressing the electron–hole recombination

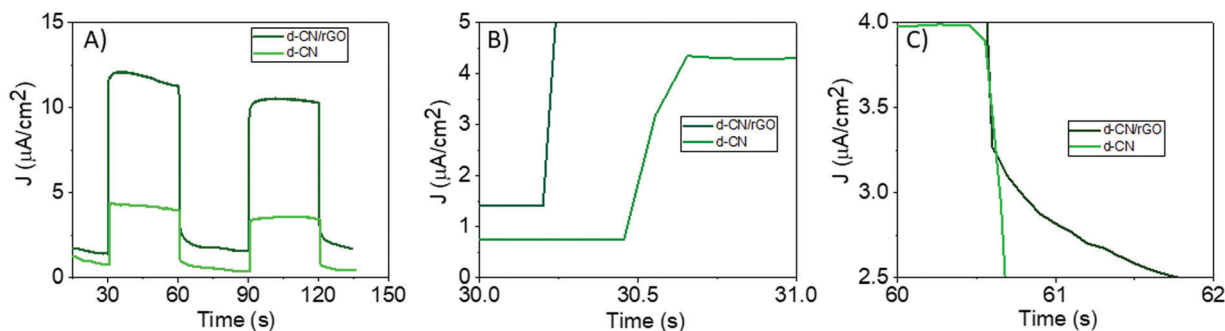


Fig. 4 Transient photocurrent response showing (A) typical on–off cycles of intermittent visible-light irradiation. (B) Rise-time dynamics and (C) decay-time dynamics of the transient photocurrent. The measurement was carried out in three-electrode configurations (working electrode, reference electrode and counter electrode) using a 0.2 M  $\text{Na}_2\text{S}$  + 0.05 M  $\text{Na}_2\text{SO}_3$  mixed aqueous solution.



in d-CN. An apparent sharp peak in the transient photocurrent signal can be qualitatively understood. When the light is switched on, photogenerated electrons move towards the bulk of the semiconductor, while the holes move toward the surface. This separation of electrons and holes causes the charging of the space charge capacitance. This means that the corresponding instantaneous displacement or charging current is not associated with charge transfer across the interface. Under continued illumination, a steady-state concentration of the free (or surface-trapped) holes builds up. Under these conditions, interfacial transfer and recombination exactly balance the rate of arrival of holes. As a result, this build-up of holes induces a recombination flux of electrons into the surface corresponding to a current of opposite sign to the hole current.

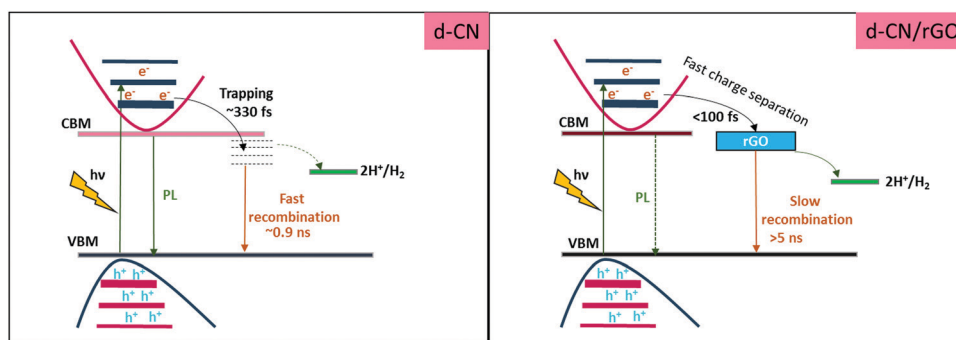
Before reaching the steady state, the rise–decay dynamics of transient photocurrent provides direct information on trapping and detrapping of charge carriers.<sup>45</sup> As can be seen in Fig. 4B, d-CN/rGO exhibits a faster increase in the steady state current than d-CN. This refers to the fact of faster detrapping of trapped-charge population in d-CN/rGO before reaching equilibrium, while the delayed rise to steady-state current observed in bare d-CN indicates the increased trap population<sup>17,45,46</sup> which induces a strong trap-mediated recombination in d-CN that results in a decreased current density (Fig. 4A). At the turn-off of the illumination, d-CN/rGO shows a long photocurrent tail for more than 750  $\mu$ s (Fig. 4C). This infers the continuous extraction of charge carriers because of suppressed recombination evident from the strong quenching of the PL intensity (Fig. 2A).

**2.4.4 rGO increases the charge carrier polarization plausibly through Fröhlich electron–phonon coupling.** Because d-CN/rGO is an indirect bandgap semiconductor, the difference in momentum in the  $k$ -space in the Brillouin zone is likely to influence the electron–phonon coupling that favorably hinders the electron–hole pair recombination. It is well-understood that electron–phonon coupling can broaden the PL spectra.<sup>47</sup> Therefore, for d-CN/rGO, we correlate the spectral broadening observed in the PL spectrum with the electron–phonon coupling. The inhomogeneous broadening of the photoluminescence spectrum arises from a combination of exciton–phonon coupling

of the lifetime-limited natural spectrum, spectral dynamics, and emission from fine-structure states due to interparticle inhomogeneities.<sup>47</sup>

Due to the lattice mismatch between rGO and d-CN, layers of rGO impart considerable strain onto the surface of d-CN with an increased likelihood of defect site formation (which is also confirmed by Raman spectroscopy measurements, Fig. S7, ESI<sup>†</sup>). Because of weak electron–acoustic phonon coupling in graphene,<sup>48</sup> lattice distortion couples excitons with higher-energy optical phonons through deformation potential interactions known as Fröhlich coupling (in which the phonons generated oscillating polarization which modulates the periodic potential that determines the electronic band energies).<sup>49–51</sup> The red-shift in the UV absorbance of d-CN/rGO is also a reflection of the scattering of incident photonic radiation through rGO.<sup>52</sup> Correspondingly, we also observed a red-shift in the PL emission spectra of d-CN/rGO. We have hypothesized that exciton–phonon coupling would dephase the electronic transition to produce phonon sidebands that would displace emission energetically from the zero-phonon transition.<sup>53,54</sup> As a result, we observed spectral shifts that corroborate the photoinduced fluctuations in local electric fields.<sup>53–55</sup> However, in the UV-vis spectra, there was only a smooth and monotonic increase of absorption with decreasing wavelength, but there were no added red-shifted excitonic peaks. Therefore, quantum confinement effects are less favorable due to the lack of sharp or well-defined spectral peaks.<sup>31</sup>

Previously it has been reported that the ground-state excitons in 2D nanosheets of rGO consist of multiple optically active electronic fine-structure states.<sup>56,57</sup> Emissions can originate from multiple electronic states each with its own phonon coupling properties. It was shown that coupling to longitudinal optical (LO) phonons had a major contribution to the spectral broadening because of an increased Fröhlich interaction due to a decreased wave function overlap between the electrons and holes (*i.e.* surface charge localization).<sup>47,58</sup> d-CN/rGO forms a quasi-type II heterostructure where the electron wave function of d-CN leaks into rGO (rGO acts as an electron sink), while the hole remains confined in d-CN. Therefore, a decrease in the overlap between the electron and hole wave functions is



**Scheme 1** Hypothetical illustration of charge transport kinetics. CBM: conduction band maximum, VBM: valence band maximum,  $h^+$ : hole,  $e^-$ : electron. Lines are not drawn to scale. Incorporation of rGO into the d-CN skeleton enhances the charge transfer 2.5-fold and slows down the recombination ~5-fold compared to bare d-CN.



expected. This will correspondingly increase the polarization of charge carriers that would enhance the coupling to LO phonons (see also ESI† note S3).<sup>48,51,59</sup>

Based on our TAS and TRPL results, here we have provided a mechanistic picture to illustrate the direction of the transport and separation of charge carriers in the d-CN/rGO system (see Scheme 1). The above TAS and TRPL results have clearly demonstrated that the addition of rGO positively influences the charge transport kinetics by suppressing the band-edge recombination, and interfacial transfer and temporal separation of electrons which lead to enhanced hydrogen production in d-CN.

### 3. Conclusion

In conclusion, this study has explored the role of rGO in enhancing the photocatalytic activity of d-CN toward hydrogen production by delving into the kinetic competition of charge separation and transport using femtosecond TAS and TRPL complemented by transient photocurrent measurements. Our findings reveal that rGO, when incorporated into d-CN, passivates traps that not only help in suppression of the band edge bimolecular recombination as well as the trap-assisted recombination, but also enhance the spatial separation of charge carriers in d-CN. As a result, d-CN/rGO increased the rate of hydrogen production by almost an order of magnitude compared to bare d-CN. By providing an insight into the kinetic competition of different photophysical processes, our study may therefore guide the pathway to ‘metal-free photocatalysts by design’.

### Experimental section

See the ESI.†

### Conflicts of interest

The authors declare no conflict of interest.

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### References

- M. Z. Rahman, M. G. Kibria and C. B. Mullins, *Chem. Soc. Rev.*, 2020, **49**, 1887–1931.
- W. J. Ong, L. L. Tan, Y. H. Ng, S. T. Yong and S. P. Chai, *Chem. Rev.*, 2016, **116**, 7159–7329.
- M. Z. Rahman, P. C. Tapping, T. W. Kee, R. Smernik, N. Spooner, J. Moffatt, Y. Tang, K. Davey and S.-Z. Qiao, *Adv. Funct. Mater.*, 2017, **27**, 1702384.
- M. Z. Rahman, K. Davey and C. B. Mullins, *Adv. Sci.*, 2018, **5**, 1800820.
- M. Z. Rahman and C. B. Mullins, *Acc. Chem. Res.*, 2019, **52**, 248–257.
- L. Kong, P. Song, F. Ma and M. Sun, *Mater. Today Energy*, 2020, **17**, 100488.
- M. Z. Rahman, T. Edvinsson and J. Gascon, *Nat. Rev. Chem.*, 2022, **6**, 243–258.
- C. Merschjann, T. Tyborski, S. Orthmann, F. Yang, K. Schwarzburg, M. Lublow, M. C. Lux-Steiner and T. Schedel-Niedrig, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 205204.
- M. Z. Rahman, Y. Tang and P. Kwong, *Appl. Phys. Lett.*, 2018, **112**, 253902.
- T. M. Clarke and J. R. Durrant, *Chem. Rev.*, 2010, **110**, 6736–6767.
- M. Rahman, G. Boschloo, A. Hagfeldt and T. Edvinsson, *Adv. Mater.*, 2020, **32**, e1905653.
- M. Rahman and K. Davey, *Phys. Rev. Mater.*, 2018, **2**, 125402.
- R. Godin, Y. Wang, M. A. Zwijnenburg, J. Tang and J. R. Durrant, *J. Am. Chem. Soc.*, 2017, **139**, 5216–5224.
- K. Appavoo, M. Liu, C. T. Black and M. Y. Sfeir, *Nano Lett.*, 2015, **15**, 1076–1082.
- Q. Wang and K. Domen, *Chem. Rev.*, 2020, **120**, 919–985.
- M. Z. Rahman, M. Batmunkh, M. Bat-Erdene, J. G. Shapter and C. B. Mullins, *J. Mater. Chem. A*, 2018, **6**, 18403–18408.
- M. Z. Rahman, J. Moffatt and N. Spooner, *Mater. Horiz.*, 2018, **5**, 553–559.
- Y. Kang, Y. Yang, L. C. Yin, X. Kang, G. Liu and H. M. Cheng, *Adv. Mater.*, 2015, **27**, 4572–4577.
- M. Rahman, H. Tian and T. Edvinsson, *Angew. Chem., Int. Ed.*, 2020, **59**, 16278–16293.
- M. Z. Rahman, J. Zhang, Y. Tang, K. Davey and S.-Z. Qiao, *Mater. Chem. Front.*, 2017, **1**, 562–571.
- Q. Xiang and J. Yu, *J. Phys. Chem. Lett.*, 2013, **4**, 753–759.
- Q. Xiang, J. Yu and M. Jaroniec, *Chem. Soc. Rev.*, 2012, **41**, 782–796.
- X. Li, R. Shen, S. Ma, X. Chen and J. Xie, *Appl. Surface Sci.*, 2018, **430**, 53–107.
- W. Lin, E. Cao, L. Zhang, X. Xu, Y. Song, W. Liang and M. Sun, *Nanoscale*, 2018, **10**, 5482–5488.
- P. Niu, L. C. Yin, Y. Q. Yang, G. Liu and H. M. Cheng, *Adv. Mater.*, 2014, **26**, 8046–8052.
- S. Yang, Y. Gong, J. Zhang, L. Zhan, L. Ma, Z. Fang, R. Vajtai, X. Wang and P. M. Ajayan, *Adv. Mater.*, 2013, **25**, 2452–2456.
- Y. Hou, Z. Wen, S. Cui, X. Guo and J. Chen, *Adv. Mater.*, 2013, **25**, 6291–6297.
- D. Li, M. B. Muller, S. Gilje, R. B. Kaner and G. G. Wallace, *Nat. Nanotechnol.*, 2008, **3**, 101–105.
- L.-C. Chen, C.-Y. Teng, C.-Y. Lin, H.-Y. Chang, S.-J. Chen and H. Teng, *Adv. Energy Mater.*, 2016, **6**, 1600719.
- P. Würfel, *Physics of Solar Cells*, Wiley-VCH, 2nd edn, 2010.
- D. A. Wheeler and J. Z. Zhang, *Adv. Mater.*, 2013, **25**, 2878–2896.
- J. Duan, S. Chen, M. Jaroniec and S. Z. Qiao, *ACS Nano*, 2015, **9**, 931–940.
- J. B. Baxter, C. Richter and C. A. Schmuttenmaer, *Annu. Rev. Phys. Chem.*, 2014, **65**, 423–447.



- 34 A. L. Stroyuk, Y. V. Panasiuk, A. E. Raevskaya and S. Y. Kuchmy, *Theor. Exp. Chem.*, 2015, **51**, 243–251.
- 35 Z. Gan, Y. Shan, J. Chen, Q. Gui, Q. Zhang, S. Nie and X. Wu, *Nano Res.*, 2016, **9**, 1801–1812.
- 36 J. Xu, M. Shalom, F. Piersimoni, M. Antonietti, D. Neher and T. J. K. Brenner, *Adv. Opt. Mater.*, 2015, **3**, 913–917.
- 37 P. Maity, N. A. Merdad, J. Yin, K. J. Lee, L. Sinatra, O. M. Bakr and O. F. Mohammed, *ACS Energy Lett.*, 2021, **6**, 2602–2609.
- 38 I. Dursun, P. Maity, J. Yin, B. Turedi, A. A. Zhumekenov, K. J. Lee, O. F. Mohammed and O. M. Bakr, *Adv. Energy Mater.*, 2019, **9**, 1900084.
- 39 J. Wang, *et al.*, *ACS Energy Lett.*, 2022, **7**, 10–16.
- 40 M. Sachs, H. Cha, J. Kosco, C. M. Aitchison, L. Francas, S. Corby, C. L. Chiang, A. A. Wilson, R. Godin, A. Fahey-Williams, A. I. Cooper, R. S. Sprick, I. McCulloch and J. R. Durrant, *J. Am. Chem. Soc.*, 2020, **142**, 14574–14587.
- 41 W. Yang, R. Godin, H. Kasap, B. Moss, Y. Dong, S. A. J. Hillman, L. Steier, E. Reisner and J. R. Durrant, *J. Am. Chem. Soc.*, 2019, **141**, 11219–11229.
- 42 W. Bi, L. Zhang, Z. Sun, X. Li, T. Jin, X. Wu, Q. Zhang, Y. Luo, C. Wu and Y. Xie, *ACS Catal.*, 2016, **6**, 4253–4257.
- 43 L. M. Peter and K. G. Ubul Wijayantha, *Chem. Phys. Chem.*, 2014, **15**, 1983–1995.
- 44 Y. H. Ng, I. V. Lightcap, K. Goodwin, M. Matsumura and P. V. Kamat, *J. Phys. Chem. Lett.*, 2010, **1**, 2222–2227.
- 45 C. R. McNeill, I. Hwang and N. C. Greenham, *J. Appl. Phys.*, 2009, **106**, 024507.
- 46 Z. Li and C. R. McNeill, *J. Appl. Phys.*, 2011, **109**, 074513.
- 47 J. Cui, A. P. Beyler, I. Coropceanu, L. Cleary, T. R. Avila, Y. Chen, J. M. Cordero, S. L. Heathcote, D. K. Harris, O. Chen, J. Cao and M. G. Bawendi, *Nano Lett.*, 2016, **16**, 289–296.
- 48 A. Laitinen, M. Oksanen, A. Fay, D. Cox, M. Tomi, P. Virtanen and P. J. Hakonen, *Nano Lett.*, 2014, **14**, 3009–3013.
- 49 A. M. Kelley, *ACS Nano*, 2011, **5**, 5254–5262.
- 50 A. M. Kelley, *J. Phys. Chem. Lett.*, 2010, **1**, 1296–1300.
- 51 C.-H. Park, F. Giustino, M. L. Cohen and S. G. Louie, *Nano Lett.*, 2008, **8**, 4229–4233.
- 52 D. P. Singh, V. Kumar, A. Kumar, R. Manohar, R. Pasricha, B. Duponchel, Y. Boussoualem, A. H. Sahraoui and A. Daoudi, *RSC Adv.*, 2017, **7**, 12479–12485.
- 53 S. A. Empedocles, R. Neuhauser, K. Shimizu and M. G. Bawendi, *Adv. Mater.*, 1999, **11**, 1243–1256.
- 54 D. T. Nguyen, C. Voisin, P. Roussignol, C. Roquelet, J. S. Lauret and G. Cassabois, *Chem. Phys.*, 2013, **413**, 102–111.
- 55 S. A. Empedocles and M. G. Bawendi, *Science*, 1997, **278**, 2114–2117.
- 56 W. E. Moerner and M. Orrit, *Science*, 1999, **283**, 1670–1676.
- 57 A. L. Efros and M. Rosen, *Annu. Rev. Mater. Sci.*, 2000, **30**, 475–521.
- 58 T. D. Krauss and F. W. Wise, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1997, **55**, 9860–9865.
- 59 E. Groeneveld and C. de Mello Donegá, *J. Phys. Chem. C*, 2012, **116**, 16240–16250.

