



Cite this: *Phys. Chem. Chem. Phys.*, 2022, 24, 488

# Quantitative structure determination of adsorbed formate and surface hydroxyls on Fe<sub>3</sub>O<sub>4</sub>(001)<sup>†</sup>

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Using the chemically specific techniques of normal incidence X-ray standing waves and photoelectron diffraction, we have investigated the dissociative adsorption of formic acid on the Fe<sub>3</sub>O<sub>4</sub>(001) surface, specifically probing the local structures of both the adsorbed formate and resulting surface hydroxyl. Using model independent direct methods, we reinforce the observations of a previous surface X-ray diffraction study that the formate molecule adsorbs with both oxygens atop octahedrally coordinated surface Fe cations and that ~60% of the formate is adsorbed in the so called *tet* site. We additionally determine, for the first time, that the surface hydroxyl species are found at the so called *int* site. This confirms previous DFT predictions and reinforces the pivotal role the surface hydroxyl plays in lifting the subsurface cation vacancy termination of the Fe<sub>3</sub>O<sub>4</sub>(001) surface.

Received 16th September 2021,  
 Accepted 11th November 2021

DOI: 10.1039/d1cp04241f

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## Introduction

Magnetite, Fe<sub>3</sub>O<sub>4</sub>, an abundant and cheap metal-oxide, has found increased scientific interest in recent decades in a host of different scientific fields. These range from the use of the mineral in both lithium ion batteries<sup>1</sup> and as an electrocatalyst in fuel cells<sup>2</sup> as well as a support material for single atom catalysts (SACs).<sup>3</sup> Many of these applications arise due to the rich catalogue of surface structures for Fe<sub>3</sub>O<sub>4</sub> which, in turn, can be attributed to its interesting redox chemistry. For example, the subsurface cation vacancy (SCV) reconstruction of the Fe<sub>3</sub>O<sub>4</sub>(001) surface is responsible for kinetically stabilising a range of different metal adatom phases, thus providing model catalytic systems with a well-defined single reactive site.<sup>3–6</sup>

Notably, for device applications, citrate or olate stabilised nanoparticles of Fe<sub>3</sub>O<sub>4</sub> have been used for tissue specific drug therapy where the negative charge of the modified nanoparticle surface can facilitate the adsorption of positively charged drugs. These drugs are then found to be released in the acidic environments of tumours, thereby creating tumour specific chemotherapy treatments.<sup>7,8</sup> In the same vein, citrate stabilised nanoparticles of Fe<sub>3</sub>O<sub>4</sub> are utilised as magnetic resonance imaging (MRI) contrast mediums, and, in many cases, are preferred over more conventional contrast mediums due to their excellent magnetic properties, tuneable biocompatibility and, importantly, their low toxicity.<sup>9,10</sup> Often, control of these

properties is achieved *via* the organic functionalisation of the Fe<sub>3</sub>O<sub>4</sub> surface, by the adsorption of appropriate surface ligands.

Though a wide range of molecular and supramolecular species are used as coatings to such magnetite nanoparticles, they commonly contain functional groups that are thought to form ionic interactions with the surface of the nanoparticle. In particular, the carboxylate functional group is widespread in surface functionalization in general,<sup>11–17</sup> and magnetite surface functionalisation specifically,<sup>18–20</sup> due to how strongly it adsorbs on the surface and its relative robustness.

Thus, understanding the interaction of carboxylate containing compounds with Fe<sub>3</sub>O<sub>4</sub> surfaces could facilitate the bottom up approach to designing and developing new organic functionalised Fe<sub>3</sub>O<sub>4</sub> materials. More generally such knowledge could also provide a fundamental insight into the structures of the Fe<sub>3</sub>O<sub>4</sub> surfaces under different conditions providing vital information for studies utilising Fe<sub>3</sub>O<sub>4</sub> in applications such as SAC, which rely heavily on such information.

The Fe<sub>3</sub>O<sub>4</sub>(001) facet, the second most energetically stable facet after the (111), reconstructs under oxidation giving the subsurface cation vacancy (SCV) termination with ( $\sqrt{2} \times \sqrt{2}$ ) *R*45° periodicity.<sup>21</sup> Gamba *et al.* investigated the interaction of formic acid with the Fe<sub>3</sub>O<sub>4</sub>(001) SCV reconstructed surface, using X-ray photoelectron spectroscopy (XPS), low energy electron diffraction (LEED), scanning tunnelling microscopy (STM), infrared reflection adsorption spectroscopy (IRRAS) and density functional theory (DFT) calculations.<sup>22</sup> XPS and IRRAS indicated that formic acid underwent dissociative adsorption at room temperature; the O–H bond of the formic acid breaks producing a surface adsorbed formate and a proton that, itself, is believed to form a surface hydroxyl. The dissociative adsorption of formic acid, according to LEED, lifted the SCV

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<sup>†</sup> Electronic supplementary information (ESI) available. See DOI: 10.1039/d1cp04241f



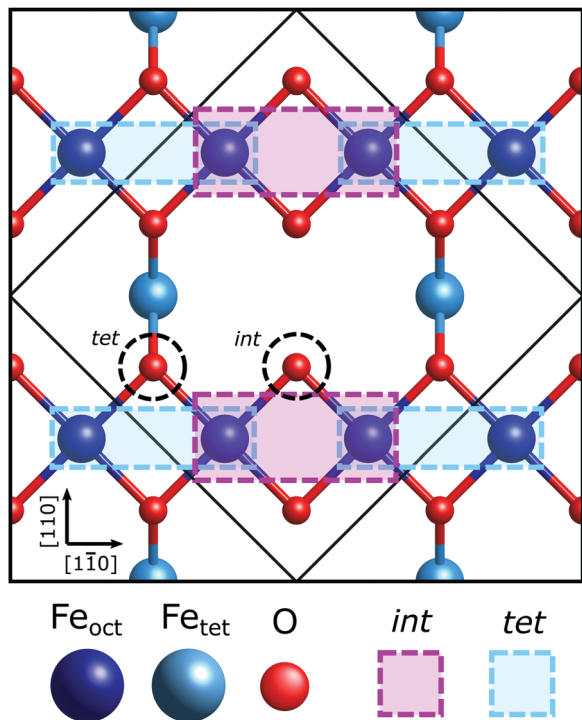


Fig. 1 Schematic of the outermost octahedral and tetrahedral layers of the bulk terminated  $\text{Fe}_3\text{O}_4(001)$  surface. The light blue boxes mark the surface  $\text{Fe}_{\text{oct}}$  cations that the  $\text{O}_{\text{HCOO}}$  bond to while in the tet site and the larger, purple boxes are for those of the *int* site. The black dotted circles mark the oxygens of the possible surface hydroxyl ( $\text{O}_{\text{OH}}$ ) sites (*tet*, *int*).

reconstruction giving a LEED pattern indicative of a bulk-like termination. A recently published surface X-ray diffraction (SXRD) study by Arndt *et al.*<sup>23</sup> has shown that the formate is bound to two surface  $\text{Fe}_{\text{oct}}$  cations through its two carboxylate oxygens,  $\text{O}_{\text{HCOO}}$ , which is consistent with how formate<sup>24–29</sup> and more broadly the carboxylate functional group,<sup>30–37</sup> bonds with most surfaces. Moreover, they were able to determine that the formate molecule preferentially binds to one of the two bidentate adsorption sites, the ‘*tet*’ adsorption site, with the formate carbon atoms coincident with the 1st subsurface  $\text{Fe}_{\text{tet}}$  in the  $[110]$  crystallographic direction, shown in Fig. 1. However, as SXRD is both chemically “blind” and insensitive to H atoms, this study was unable to probe the location of the surface hydroxyl. Furthermore, SXRD analyses is model dependent, requiring prospective structures to be theoretically modelled and compared against experimental results, thus if the correct structure is not modelled, it will not be found. Understanding the adsorption site of this hydroxyl species has significant importance in understanding how the SCV reconstruction is lifted upon hydroxylation, which is of interest not only to fundamentally understand the chemistry of  $\text{Fe}_3\text{O}_4$ , but also for investigations concerning the use of  $\text{Fe}_3\text{O}_4$  as a support for SACs, where transient hydroxylation of the support surface will occur as part of the catalytic cycle.

As such, in this work we exploit the chemical specificity of energy-scanned photoelectron diffraction (PhD) and normal incidence X-ray standing wave (NIXSW) techniques to

determine the adsorption geometry of the adsorbed formate and the location of  $\text{O}_{\text{OH}}$ . We determine the adsorption site of formate on  $\text{Fe}_3\text{O}_4(001)$  to sub-ångström resolution, confirming prior studies, and identify the preferred adsorption site of the surface hydroxyl. We determine that the surface hydroxyl does indeed occupy the interstitial site (*int*) of the surface. Ultimately, this experimentally reinforces the important role the surface hydroxyl plays in the mechanism for the lifting of the SCV reconstruction.

## Experimental details

All measurements were conducted at 300 K in the permanent ultra-high vacuum (UHV –  $\sim 3 \times 10^{-10}$  mbar) end-station on the I09 beamline<sup>38</sup> at the Diamond Light Source. Beamline I09 utilises two separate undulators which are monochromated by a double Si(111) crystal monochromator and a plane grating monochromator. These two separate lines provide simultaneous access to both ‘hard’ and ‘soft’ X-ray energies. Specifically, we have used incident photon energies of 650 eV and 435 eV for all the O 1s and C 1s soft X-ray photoelectron spectroscopy (SXPS) measurements, respectively. A photon energy range of 600–890 eV was used for the O 1s PhD measurements. For the NIXSW, ranges of  $\sim 2950$ – $2960$  eV,  $\sim 2950$ – $2960$  eV and  $\sim 4170$ – $4180$  eV were used for the  $\{113\}$ , (004) and (044) reflections, respectively. The absolute binding energy scale of all XP spectra were calibrated by subsequent measurements of the Fermi edge, which defined the binding energy scale origin.

All photoelectron spectra were acquired using a VG Scienta EW4000 HAXPES hemispherical electron analyser (angular acceptance range  $\pm 28^\circ$ ) mounted perpendicular to the incident radiation and in plane with the polarisation of the incident photon (linear-horizontal). All peaks in the XP spectra were fitted using a numerical convolution of a Gaussian and a Lorentzian peak shape. Within each core-level the peaks were allowed to take different fitted Gaussian widths but the same fitted Lorentzian width. For all peaks in the XPS data, the binding energy was a fitting parameter except for the bulk O 1s core-level peak whose binding energy was fixed at the literature value of 530.0 eV.<sup>39</sup>

A  $\text{Fe}_3\text{O}_4(001)$  single crystal ( $\pm 0.1^\circ$ , purchased from and polished by the Surface Preparation Laboratory and PI-KEM) was cleaned *in situ* via ten cycles of sequential sputtering ( $\text{Ar}^+$ , voltage: 1 keV, emission current: 3 mA, 10 min) and annealing ( $\sim 600^\circ\text{C}$ , 10 min). The annealing was alternated between annealing in UHV and in an oxygen partial pressure of  $1 \times 10^{-6}$  mbar with the final anneal in an oxygen partial pressure. The prepared samples showed a sharp ( $\sqrt{2} \times \sqrt{2}$ )  $R45^\circ$  LEED pattern indicating the presence of SCV surface reconstruction.

Formic acid sourced from Sigma Aldrich was purified by several freeze–pump–thaw cycles. The clean  $\text{Fe}_3\text{O}_4(001)$  surface was exposed to  $5 \times 10^{-8}$  mbar of formic acid for 200 seconds ( $\sim 10$  L, where 1 L is  $1 \times 10^{-6}$  mbar s).

The details of the NIXSW and PhD techniques are given in the electronic ESI† (Sections 1 and 2).



## Results

### SXPS

Fig. 2a shows the peak fitted C 1s core-level SXPS data after exposure to formic acid. Three species are observed and assigned as a carbon shake up feature ( $C_S$ ) at 289.4 eV, carbon from adsorbed formate ( $C_{HCOO}$ ) at 289.1 eV and adventitious carbon ( $C_{ad}$ ) at 285.0 eV. Intact formic acid is not expected on the surface as, in the previous work by Gamba *et al.*,<sup>22</sup> the C=O and C–OH stretching modes were absent above 190 K.

The O 1s SXP spectrum, shown in Fig. 2b, exhibits three features: bulk oxide, ( $O_{bulk}$ ) at 530.0 eV, a surface hydroxyl oxygen ( $O_{OH}$ ) at 531.7 eV, and adsorbed formate, ( $O_{HCOO}$ ) at 531.7 eV. The assignments are in close agreement with those of Gamba *et al.*<sup>22</sup> as well as for carboxylates on other metal–oxide surfaces.<sup>40,41</sup> These SXPS results were used to fit the corresponding spectra for both the PhD and NIXSW measurements. A direct comparison of the O 1s region both before and after dosing formic acid is given in Fig. S1 in Section 3 (ESI<sup>†</sup>).

### NIXSW – adsorption heights

Fig. 3 shows the fitted photoelectron intensity profiles of the (004) reflection for the  $C_{HCOO}$ ,  $O_{HCOO}$  and  $O_{OH}$  species after saturating the SCV surface with formic acid. Also inset to the figures are the values of the corresponding coherent fraction,  $f^{004}$ , and position,  $P^{004}$ , for each of the fits.

Both the  $O_{HCOO}$  and  $O_{OH}$  photoelectron profiles (Fig. 3a and b) are similar indicating that both these species occupy near identical positions relative to the bulk (004) planes. This understanding is reinforced by the similarity in the fitted coherent positions for the  $O_{OH}$  and  $O_{HCOO}$  species  $P_{O_{OH}}^{004} = 0.47 \pm 0.04$  and  $P_{O_{HCOO}}^{004} = 0.52 \pm 0.02$ , respectively. Since, when fitting the NIXSW data, the origin of the substrate's unit cell was defined to be centred on a tetrahedrally coordinated Fe cation layer, these values of  $P^{004}$  place either species close to coincidence with a bulk or bulk-projected octahedrally coordinated Fe cation layer, *i.e.*  $P^{004} \approx 0.5$ . The corresponding heights relative to a bulk-like

octahedrally terminated surface are  $2.14 \pm 0.04 \text{ \AA}$  and  $-0.04 \pm 0.04 \text{ \AA}$  for the  $O_{HCOO}$  and  $O_{OH}$  atoms respectively. Note that the height difference between the  $O_{HCOO}$  and  $O_{OH}$  atoms is similar to the (004) bulk spacing as the  $O_{OH}$  atom is assumed as being in the surface and the  $O_{HCOO}$  atom is assumed as existing above the surface at a projected (004) lattice plane.

The photoelectron intensity profile for the  $C_{HCOO}$  species (Fig. 3c) is clearly different from that of the oxygen species', identifying that it sits at a significantly different position relative to the bulk Bragg diffraction planes. The fitted coherent position,  $P_{C_{HCOO}}^{004} = 0.74 \pm 0.08$ , places the C atom  $0.6 \pm 0.2 \text{ \AA}$  above the  $O_{HCOO}$  atoms as is expected for a formate molecule adsorbed in a bidentate configuration.

The associated  $f^{004}$  for the  $O_{OH}$ ,  $O_{HCOO}$  and  $C_{HCOO}$  species, respectively, are  $0.9 \pm 0.1$ ,  $0.78 \pm 0.05$  and  $0.7 \pm 0.2$ . The coherent fraction for the atoms of the formate ( $C_{HCOO}$  and  $O_{HCOO}$ ) are significantly lower than that of the bulk O and the  $O_{OH}$  species. Were these coherent fractions the result of thermal vibrations, they would suggest a root mean square (rms) vibrational amplitude of  $0.28 \text{ \AA}$  for the  $C_{HCOO}$  atoms and  $0.25 \text{ \AA}$  for the  $O_{HCOO}$  atoms. This is confirmed by the PhD results presented below which were fitted with a rms vibrational amplitude of approximately  $0.3 \text{ \AA}$ . This could indicate a particularly large vibrational amplitude normal to the surface for the adsorbed formate.

### NIXSW – real space imaging

The photoelectron yield profiles for the other measured reflections ((044), (113), ( $\bar{1}$ 13), (131) and ( $\bar{3}$ 11)) for the carbon and oxygen species can be found in Fig. S2 and S3 in Section 4 (ESI<sup>†</sup>) and their corresponding coherent fractions and positions are given in Table S1 in Section 4 (ESI<sup>†</sup>). The results from these reflections, along with the (004) data, were used to produce three-dimensional real space density maps for each species *via* a Fourier expansion.<sup>42</sup> Fig. 4 shows the real space density map for the  $C_{HCOO}$  species at its corresponding (004) height above the unit cell. Note that, as the number of Fourier components in the expansion is non-infinite,

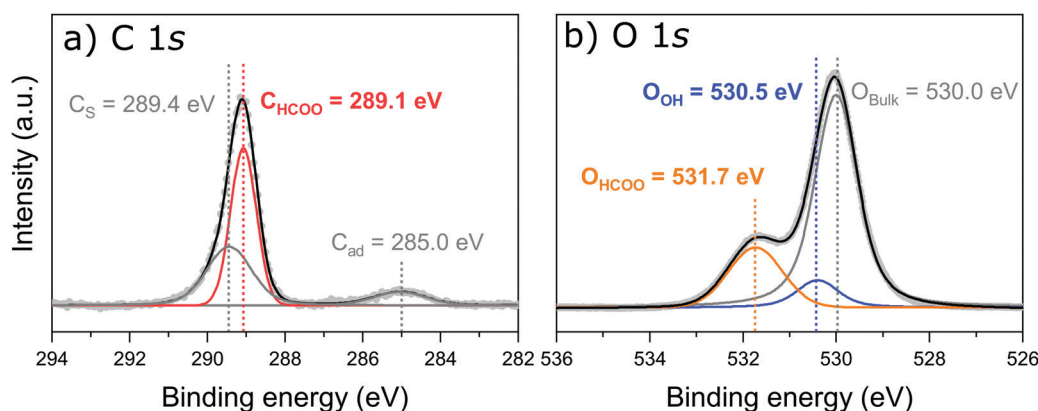


Fig. 2 Peak fitted soft XPS spectra after exposing the  $Fe_3O_4(001)$  surface to formic acid. (a) The C 1s core-level, measured with a photon energy of 450 eV, fitted with three peaks; a carbon shakeup feature,  $C_S$ , carbon from adsorbed formate,  $C_{HCOO}$ , and adventitious carbon,  $C_{ad}$ . (b) The O 1s core-level, measured with a photon energy of 680 eV, with three identified species; oxygen from the bulk metal–oxide,  $O_{Bulk}$ , formate oxygen,  $O_{HCOO}$ , and a surface hydroxyl oxygen,  $O_{OH}$ .



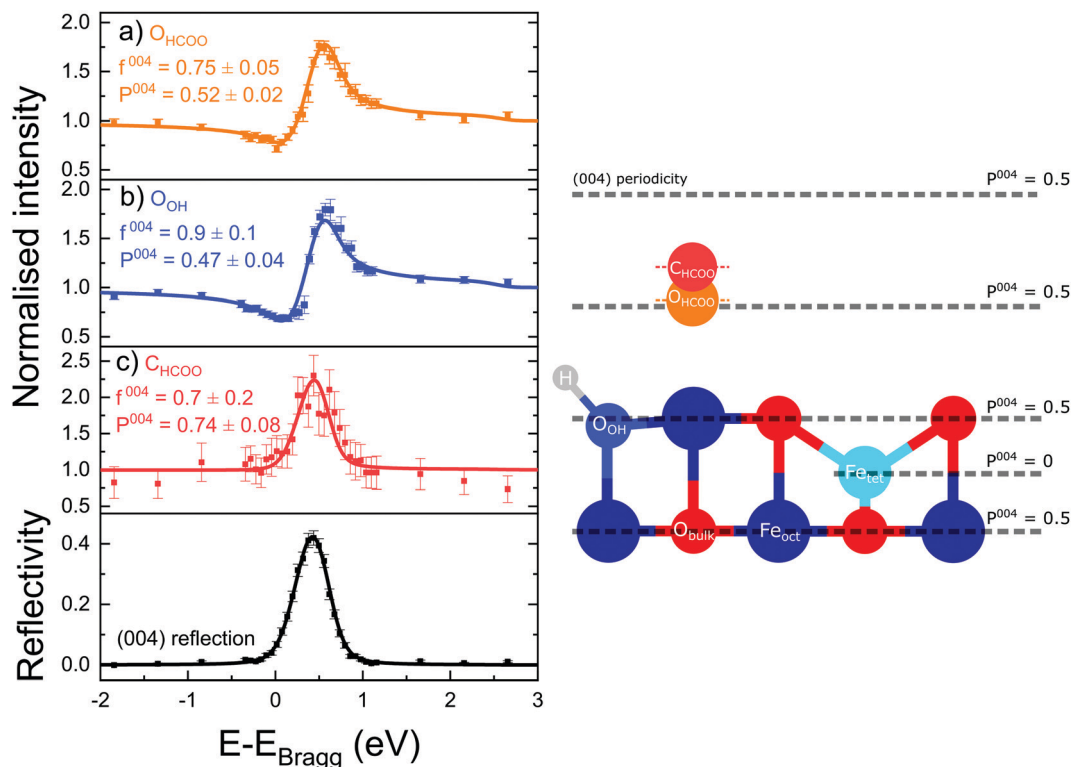


Fig. 3 Depicted is the (004) reflection data and the corresponding fitted photoelectron profiles for the (a)  $\text{O}_{\text{HCOO}}$ , (b)  $\text{O}_{\text{OH}}$  and (c)  $\text{C}_{\text{HCOO}}$  species. The coherent positions,  $p^{004}$ , and coherent fractions,  $f^{004}$ , for each fit are also given. The  $\text{O}_{\text{HCOO}}$ ,  $\text{O}_{\text{OH}}$  and  $\text{C}_{\text{HCOO}}$  positions are shown schematically on the right with respect to a bulk truncated  $\text{Fe}_3\text{O}_4(001)$  surface (no surface relaxations are shown). The  $\text{O}_{\text{HCOO}}$  and  $\text{O}_{\text{OH}}$  coherent positions are very similar though the  $\text{O}_{\text{OH}}$  atom is assumed to be coincident with a lattice plane in the surface. These results are consistent with the formate bidentate bonding to the surface.

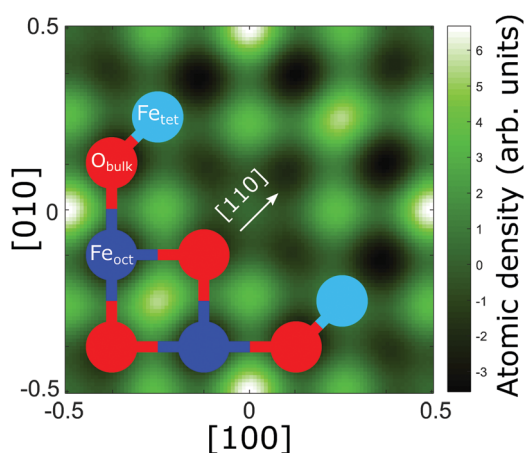


Fig. 4 The NIXSW real space image for the  $\text{C}_{\text{HCOO}}$  species taken at its corresponding (004) adsorption height ( $p_{\text{C}_{\text{HCOO}}}^{004} = 1.74$ ) above the  $\text{Fe}_3\text{O}_4(001)$  surface. For comparison, overlaid is half of the octahedrally ( $p^{004} = 0.5$ ) and tetrahedrally ( $p^{004} = 0.0$ ) coordinated layers of the bulk  $\text{Fe}_3\text{O}_4$  unit cell. The colour scale represents the probability of atomic density at that site, with white representing maximal probability, black minimal probability.

the presence of artefacts in these real space images is to be expected. Thus, we must be careful in deciding which maxima in

the real space images are real, and which are not. However, in the density map of the C species, there are clear maxima at the sites halfway between the lateral positions of the  $\text{Fe}_{\text{oct}}$  atoms along the [110] direction and a clear global maximum at the *tet* site. Therefore, we immediately confirm the results of the SXRD study of Arndt *et al.*<sup>23</sup> in a model independent manner.

In the SXRD results of Arndt *et al.*<sup>23</sup> a partial co-occupation of the *int* site ( $15 \pm 10\%$ ) was observed. Our NIXSW results presented here favour adsorption of the formate molecule in the *tet* site (where the global maximum density occurs in Fig. 4), but do also suggest that partial occupation of the *int* site (where the second highest density occurs in Fig. 4) cannot be excluded. As such, we undertook a fitting of the total  $\text{C}_{\text{HCOO}}$  unit cell density with the calculated total unit cell densities for a formate molecule in the *tet* and *int* sites with the aim of determining the percentage occupancy of both sites. A full description of this procedure is given in Section 5 (ESI<sup>†</sup>) and the corresponding example images for the calculated *tet*, *int* and best fit densities at the  $\text{C}_{\text{HCOO}}$  (004) height above the unit cell are also shown in the Section 5 (Fig. S4, ESI<sup>†</sup>). The best fit comprised a model with  $62 (\pm 5)\%$  of  $\text{C}_{\text{HCOO}}$  in the *tet* site and  $38 (\pm 5)\%$   $\text{C}_{\text{HCOO}}$  in the *int* site.

The imaging for the  $\text{O}_{\text{HCOO}}$  species (see Fig. S5a in Section 6, ESI<sup>†</sup>) have peak intensities in sites directly atop  $\text{Fe}_{\text{oct}}$  and O atoms in the (004) layer below the  $\text{O}_{\text{HCOO}}$ . This imaging corresponds to the  $\text{O}_{\text{HCOO}}$  being found atop the surface  $\text{Fe}_{\text{oct}}$



atoms with the extra maxima above the surface O atoms being due to the symmetry of the bulk crystal. This adsorption site for the  $O_{\text{HCOO}}$  agrees well with the sites identified by Arndt *et al.*<sup>23</sup> The imaging of the  $O_{\text{OH}}$  species (Fig. S5b in Section 6, ESI†) places it in a bulk oxygen site on the surface. However, it is not possible, solely from the NIXSW, to differentiate between the *int* or *tet* sites for the  $O_{\text{OH}}$  adsorption site.

### PhD – qualitative analysis

Though it is necessary to perform multiple scattering calculations in order to obtain quantitative results for PhD data, a direct inspection of the data can provide a qualitative insight into the adsorption structure. The O 1s PhD spectra for the  $O_{\text{Bulk}}$ ,  $O_{\text{OH}}$  and  $O_{\text{HCOO}}$  species in both the [100] and [110] azimuthal directions are shown in Fig. 5. The presence of strong modulations at low emission angles, with respect to normal emission (NE), for both the  $O_{\text{OH}}$  and  $O_{\text{HCOO}}$  species indicates that these species are located atop or near-atop strongly scattering surface atoms, *i.e.* Fe atoms.

More generally, the modulations present in the  $O_{\text{OH}}$  and  $O_{\text{HCOO}}$  spectra are similar, though with key differences, to those

found for the  $O_{\text{Bulk}}$  demonstrating that both the  $O_{\text{OH}}$  and  $O_{\text{HCOO}}$  sit at bulk or bulk-like oxygen sites. Whereas the  $O_{\text{Bulk}}$  and  $O_{\text{OH}}$  modulations remain well correlated with one another with increasing photoelectron kinetic energy, the period of those modulations for the  $O_{\text{HCOO}}$  species become longer particularly for the emission angles close to NE. This difference suggests that the  $d(\text{Fe}-O_{\text{HCOO}})$  bond length is shorter than that of the  $d(\text{Fe}-O_{\text{Bulk}})$  or  $d(\text{Fe}-O_{\text{OH}})$ .

The greatest difference, between the  $O_{\text{OH}}$  and  $O_{\text{Bulk}}$  modulations, occur for the spectra measured at an emission angle of  $10^\circ$  off NE along the [110] azimuth direction. Notably, the  $O_{\text{Bulk}}$  modulations are much stronger, suggesting the existence of a strong scattering atom in this direction for the bulk oxygen atoms which is not situated adjacent to the  $O_{\text{OH}}$ . The [110] azimuthal direction points down the axis of the surface  $O\text{-Fe}_{\text{tet}}$  bonds and so the weak modulations found for  $O_{\text{OH}}$  in this direction could indicate that the  $O_{\text{OH}}$  sits in the *int* surface site, not adjacent to the  $\text{Fe}_{\text{tet}}$  atoms, which is confirmed by the quantitative fitting below.

### PhD – quantitative analysis

Table 1 gives the *R*-factors of all four single-site optimised structures on a bulk terminated surface. These *R*-factors were determined *via* optimising fits to the experimental modulations using a particle swarm optimisation (PSO) algorithm.<sup>43</sup> See the ESI† for further details. Other surface terminations than bulk terminated were also considered (see the Section 7, ESI†), however the bulk terminated surface, as determined by Arndt *et al.*<sup>23</sup> was found to have the best agreement with the experimental data (see the Section 8, ESI†). The single structure that best fits the PhD modulations has the formate in the *tet* position and with the  $O_{\text{OH}}$  in the *int* position situated above a subsurface  $\text{Fe}_{\text{oct}}$  cation. The fits of the optimised structure to the experimental modulations for both the  $O_{\text{OH}}$  and  $O_{\text{HCOO}}$  species are given in Fig. 6 and the structure is shown in Fig. 7. After parameter optimisation (see the Section 9, ESI†), this model has an *R*-factor = 0.22 (variance of 0.02). After considering substrate layer relaxations, the height of the carboxylate oxygens above the surface in this optimised structure is consistent with the height determined by NIXSW. Moreover, the bond length between the  $O_{\text{HCOO}}$  and  $\text{Fe}_{\text{oct}}$  ( $d_{O_{\text{HCOO}}-\text{Fe}_{\text{oct}}}$ ) atoms, and the intermolecular  $O_{\text{HCOO}}\text{-}O_{\text{HCOO}}$  distance ( $d_{O_{\text{HCOO}}\text{-}O_{\text{HCOO}}}$ ) for this best fit optimised structure are consistent with the results of Arndt *et al.*<sup>23</sup> (see Table 2), and also agree well with other structural studies of carboxylates on both metal<sup>25–29</sup> and metal-oxide<sup>24</sup> surfaces. Further structural parameters are given in Table S3 in Section 8 (ESI†). The bond length between  $O_{\text{OH}}$  and the  $\text{Fe}_{\text{oct}}$

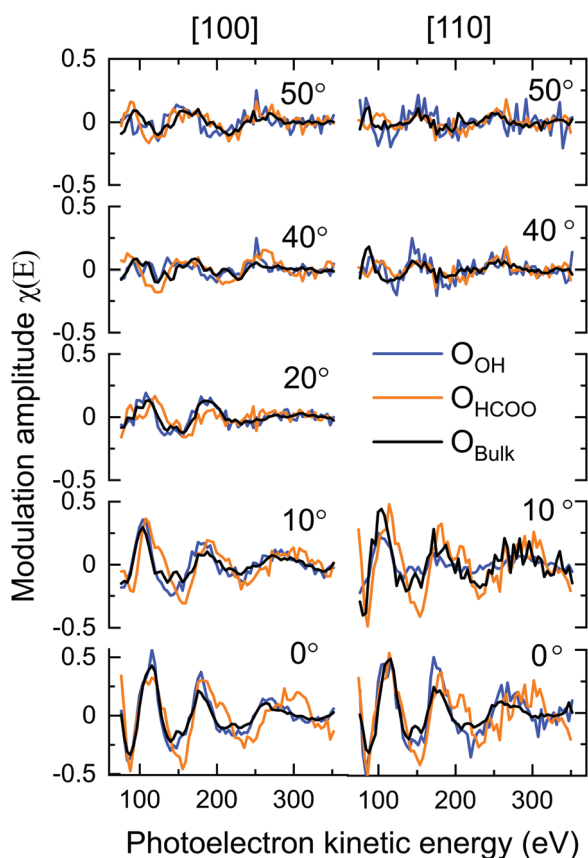


Fig. 5 The experimental modulations for the 18 beams used in the PhD analysis for the  $O_{\text{HCOO}}$  (orange),  $O_{\text{OH}}$  (blue) and  $O_{\text{Bulk}}$  (black) species. Note, only the  $O_{\text{HCOO}}$  and  $O_{\text{OH}}$  species were used as emitters in the multiple scattering calculations and the  $O_{\text{Bulk}}$  beams are provided here for comparison only. Polar emission angles are given with respect to normal emission. The  $O_{\text{HCOO}}$  and  $O_{\text{OH}}$  modulations are very similar to the those of the  $O_{\text{Bulk}}$  indicating that the  $O_{\text{HCOO}}$  and  $O_{\text{OH}}$  species sit in bulk or bulk projected positions on the surface.

Table 1 *R*-Factors, with variance in the last significant figure in brackets, of all four optimised structures bulk terminated surface

HCOO position	$O_{\text{OH}}$ position	<i>R</i> factor
<i>tet</i>	<i>int</i>	0.22(2)
<i>tet</i>	<i>tet</i>	0.32(3)
<i>int</i>	<i>int</i>	0.30(3)
<i>int</i>	<i>tet</i>	0.36(3)



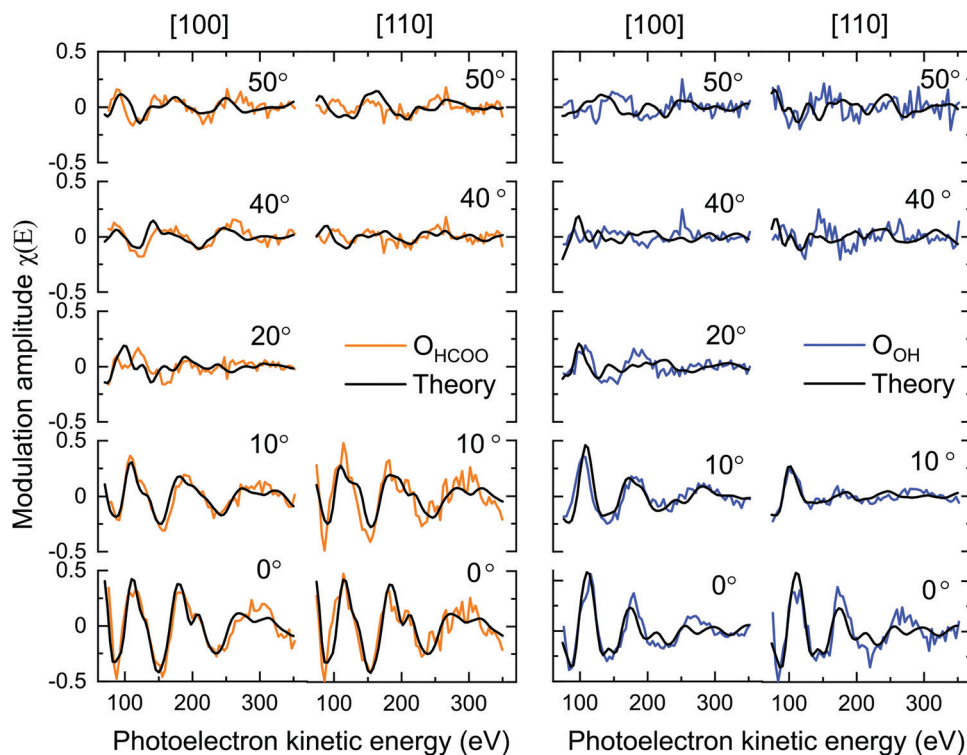


Fig. 6 Fits to the PhD experimental modulations along the two azimuthal directions ([100] and [110]). The polar angle is provided for each beam.

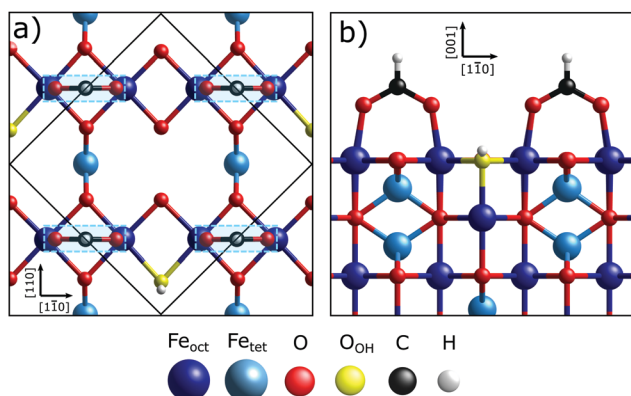


Fig. 7 (a) Plan view and (b) side view (along the [110] direction) for the best fit bulk structure. The formate is in the *tet* position, as marked by the overlaid light blue boxes, while the  $O_{OH}$  atom is in the *int* position.

directly below it ( $d_{O_{OH}-Fe_{oct}}$ ) is also included in Table 2 and is discussed below.

The prior work by Arndt *et al.*<sup>23</sup> and the NIXSW results point to a co-occupation of the *tet* and *int* sites by the formate. For the PhD, the percentage occupation of the formate in the *tet* and *int* sites was determined by undertaking domain averaged PSO optimisations of the PhD measurements. In such a calculation, an additional structural parameter determining the fraction of occupation for one of the adsorption sites was optimised. These PhD calculations found that  $70 \pm 50\%$  of the formate occupies the *tet* site which agrees well with both the NIXSW results

Table 2 Key structural parameters for the best fit structure of our PhD study compared to the SXRD study of Arndt *et al.* In particular the bond length between  $O_{HCOO}$  and  $Fe_{oct}$  ( $d_{O_{HCOO}-Fe_{oct}}$ ), the bond length between  $O_{OH}$  and the  $Fe_{oct}$  directly below it ( $d_{O_{OH}-Fe_{oct}}$ ) and the intermolecular  $O_{HCOO}-O_{HCOO}$  distance ( $d_{O_{HCOO}-O_{HCOO}}$ ) are given

Model	$d_{O_{HCOO}-Fe_{oct}}$ (Å)	$d_{O_{OH}-Fe_{oct}}$ (Å)	$d_{O_{HCOO}-O_{HCOO}}$ (Å)
PhD	1.98 ( $\pm 0.05$ )	2.2 (+0.2/−0.1)	2.4 ( $\pm 0.1$ )
SXRD <sup>23</sup>	1.96 ( $\pm 0.07$ )	—	2.51 ( $\pm 0.05$ )

above, and Arndt *et al.*<sup>23</sup> Moreover, an improvement of the *R*-factor to 0.19 (0.02) is seen when including domain averaging over both of the sites.

## Discussion

In this study the adsorption site of the formate has been directly probed in a model independent manner by NIXSW real space imaging, finding that a majority of the formate,  $62 \pm 5\%$ , is adsorbed in the *tet* site on the surface coincident with the 1st layer  $Fe_{tet}$  atoms. These results confirm the model dependent results of Arndt *et al.*<sup>23</sup> ( $75 \pm 10\%$ ) and our PhD optimisations ( $70 \pm 50\%$  for the bulk terminated structure). It could be that the preference for the *tet* site is patterned by the SCV reconstruction with the mechanism of its lifting consistently causing the formate to be located in the *tet* site on the surface. This would not be hard to rationalise as the reactivity of the  $Fe_3O_4(001)$  surface to various adsorbates is



dictated by the atomic and electronic structure of the SCV reconstruction. Notably, the SCV reconstruction contains two highly reactive oxygen anions which DFT predicts have charges of  $-1$ ,<sup>21</sup> and it is likely that the dissociative adsorption of the formic acid preferentially occurs at these sites. These reactive oxygen anions act as H-acceptors during the dissociation of the formic acid's O–H bond, accepting the  $H^+$  while the resulting  $HCOO^-$  binds to surface  $Fe_{oct}$  cations.

That  $H^+$  dissociation must occur at the *int* site, may indicate the origin of the formate's adsorption site preference at the *tet* site. Intact formic acid molecules that transiently bind to the *tet* site would have their carboxylic H atom directed towards an *int* site; transient adsorption of formic acid in the *int* site would, instead, direct this H atoms towards a *tet* site. Previous STM measurements (Gamba *et al.*<sup>22</sup>) have indicated that formate is mobile on the surface. Thus, while all formate molecules may adsorb onto the surface at a *tet* site, the mobility of the molecule would result in a stochastic mixture of adsorption in the two sites.

The formate molecule is found to adsorb on the surface with two different lateral arrangements, as indicated by STM.<sup>22</sup> These were assigned to a  $(1 \times 1)$  and  $(2 \times 1)$  periodicities on the surface, with the  $(2 \times 1)$  arrangement consisting of two apparent formate molecules offset along the  $[110]$  direction on neighbouring  $Fe_{oct}$  rows on the surface. This would be consistent with one formate molecule in an *int* site and one in a *tet* site. It is important to note that no such overlayer mesh has been observed in LEED studies, likely because the islands sizes of the  $(2 \times 1)$  periodicity are too small. Were the two periodicities of approximately equal coverage on the surface, as STM indicates, then the relative coverage of *tet* to *int* would be 75:25. The observed  $\sim 60:40$  ratio implies a 1:2 ratio of  $(1 \times 1):(2 \times 1)$ , such a high amount of  $(2 \times 1)$  on the surface would not be consistent with the STM studies nor the lack of additional spots in the LEED. Therefore, it is possible that some of the formate in the  $(1 \times 1)$  periodicity are occupying *int* sites.

The only other quantitative surface structural study into the adsorption of a hydroxyl species on a metal oxide surface is OH on  $TiO_2(110)$ .<sup>44</sup> In this study the  $d(M-O_{OH})$  was found to be  $1.97 \pm 0.05$  Å, corresponding well with the bulk Ti–O bond length. Here, however,  $d(Fe-O_{OH})$  was found to be systematically longer than the bulk  $d(Fe-O)$  distance of 2.06 Å. This corroborates with bulk bond length differences between  $d(Fe-O)$  and  $d(Fe-O_{OH})$  in iron oxide-hydroxide minerals goethite ( $\alpha-FeO(OH)$ ),<sup>45</sup> akageneite ( $\beta-FeO(OH)$ )<sup>46</sup> and lepidocrocite ( $\lambda-FeO(OH)$ ),<sup>47</sup> all of which contain both Fe–O and Fe– $O_{OH}$  bonds. Additionally, Fe cation complexes containing either a hydroxy ligand, OH, or an oxo ligand, O, are also found to follow this trend with the  $d(Fe-O_{OH})$  bond length consistently being found to be longer.<sup>48–50</sup>

Within this work we have determined the  $O_{OH}$  to occupy an *int* site. This is consistent with DFT calculations performed by Arndt *et al.*<sup>23</sup> and was also the site predicted by Bourgund *et al.*<sup>51</sup> for native surface hydroxyls on the  $Fe_3O_4(001)$  surface due to the adsorption of background gas phase hydrogen. Moreover, DFT calculations into the formation of water

monolayers on the  $Fe_3O_4(001)$  surface indicate that these monolayers are initiated by the formation of surface hydroxyls which occupy the *int* site.<sup>52</sup> Our results are the first structural determination of hydroxyl occupation in the *int* site, strongly supporting the DFT calculations of these prior works. These results are also consistent with the behaviour of deposited metal adatoms on the SCV reconstruction, which are found to adsorb at the two highly reactive surface oxygen anions of charge  $-1$ .<sup>4</sup>

## Conclusions

In this study we have used the chemical specificity of the PhD and NIXSW techniques to determine the adsorption geometry of both the formate and surface hydroxyl formed during room temperature dissociation of formic acid on  $Fe_3O_4(001)$ . Using model independent direct methods, we have confirmed previous assignments<sup>22,23</sup> of the formate adsorption site as bidentate bonding to the surface through its two  $O_{HCOO}$  atoms to two surface  $Fe_{oct}$  cations. Additionally, we find that a majority of the formate,  $62 \pm 5\%$ , is in the *tet* adsorption site (see Fig. 1) with a minority of the formate,  $38 \pm 5\%$ , in the *int* adsorption site. This is consistent with the PhD measurements which find that  $70 \pm 50\%$  of the formate is in the *tet* site on a bulk-like substrate.

We have also determined experimentally, for the first time, the adsorption site of the surface hydroxyl as the *int* site corroborating several DFT calculations and STM measurements into the site of the hydroxyl species.<sup>23,51,52</sup> Through determining experimentally the surface hydroxyl adsorption site we have reinforced the pivotal role that hydroxyl plays in the lifting of the SCV reconstruction. The preference of the surface hydroxyl to be situated in a bulk-like *int* site likely drives the  $Fe_{int}$  cation into the first subsurface cation vacancy, initiating the destabilisation of the SCV reconstruction to cation diffusion from the bulk. This study adds to the growing understanding of how hydroxyl forming species interact with the  $Fe_3O_4(001)$  surface, and particularly how they influence the chemistry of the SCV reconstruction. This work is therefore of interest in many fields including those concerning the functionalisation of oxide nanoparticles for a range of applications but also within the field of catalysis where  $Fe_3O_4$  is utilised as a promising substrate material for single atom catalysts.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

We would like to acknowledge Diamond Light Source for the award of beam time (SI17635-1). The computing resource was provided by the STFC Scientific Computing Department's SCARF cluster. PTPR was supported by the Advance Characterisation of Materials (ACM) CDT.



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