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9, 17099Porphyrin based metal–organic frameworks:
highly sensitive materials for optical sensing
of oxygen in gas phase†Tobias Burger,^a Christian Winkler,^b Irene Dalfen,^a Christian Slugovc^{b,*c} and
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The optical oxygen sensing capabilities of the porphyrin-based metal–organic frameworks, PCN-224, Pt(II)PCN-224 and Pd(II)PCN-224 were investigated. The bimolecular quenching constants (k_q) of 37 000 (PCN-224), 6700 (Pd(II)PCN-224) and 3900 Pa^{−1} s^{−1} (Pt(II)PCN-224) were found and reveal an exceptionally high oxygen-permeability for these materials. Fast gas transport within the network, large pore sizes, and electronic and spatial isolation of the porphyrin indicator in the framework are held responsible for the unprecedentedly high k_q values. PCN-224 shows 6.7 ns fluorescence lifetime and the fluorescence in air is quenched 4.2-fold. The metal–organic frameworks based on phosphorescent Pt(II) and Pd(II) porphyrins possess significantly longer decay times of 18.6 and 390 μs, respectively, and are suited to detect oxygen in trace and ultra-trace ranges with limits of detection of 1 and 0.015 Pa, respectively. Apart from free-standing crystals, also metal–organic frameworks supported on different fibrous substrates (poly(acrylonitrile) nanofibers, glass fibres), and flat substrates (TLC silica-gel, poly(amide) filter) were prepared in order to provide oxygen sensor materials of practical use. Electrospun and thermally treated poly(acrylonitrile) nanofibers were found to be particularly favourable and the resulting composite material exhibited the same sensitivity as the free crystals. All sensing materials show reversible cross-talk to humidity at levels up to 53% relative humidity but demonstrate a drastic decrease of oxygen sensitivity at high humidity levels and when exposed to water. A decrease of the quenching constant with rising temperature indicates an important role of surface-adsorbed oxygen in the quenching process.

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Introduction

Widely applied optical oxygen sensors rely on quenching of the luminescence of an indicator by molecular oxygen.¹ Although the state-of-the-art sensors are well suited for most applications and some are commercially available, there is still considerable interest for further improvements in applications like trace or ultra-trace sensing² or measurements under harsh conditions

(high pressures, temperatures, presence of organic solvents *etc.*).^{3,4} Conventional sensors are made in the form of planar foils or fibre-optic setups and are commonly prepared by immobilization of molecular indicators into an oxygen-permeable matrix that acts as a support and solvent for the dye and a permeation-selective membrane. This is also true for nanoparticle-based oxygen probes used in biological and medical research,⁵ but indicators conjugated to proteins⁶ and peptides^{7,8} as well as dendrimeric probes proved very successful alternatives.⁹ The performance of optical oxygen sensors is mainly determined by the properties of the indicator (spectral properties, photostability, luminescence decay time) and the matrix (chemical stability, oxygen permeability, permeation selectivity). Compatibility of both materials is important since aggregation of the indicator negatively affects the sensor performance. The sensitivity of optical oxygen sensors is mainly governed by two parameters: luminescence decay time of the indicator and oxygen permeability of the matrix.¹⁰ Considerably long decay times are required to achieve sufficient sensitivities in common matrices which explains why fluorescent dyes, popular in the

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† Electronic supplementary information (ESI) available: The supporting information contains preparations of the precursors, a detailed description of the PXRD experiments and results, SEM images of free crystals and the immobilized MOFs, nitrogen adsorption isotherms of the three MOFs, details on custom-made adaptors for oxygen quenching studies, fit parameters for the Stern–Volmer plots for MOFs and reference samples. See DOI: 10.1039/d1tc03735h

early era of oxygen sensor development,^{1,11} are rarely used nowadays. Among numerous phosphorescent indicators explored, metalloporphyrins are probably the most popular representatives.^{11,12} Palladium(II) and platinum(II) are almost exclusively used as central atoms; their porphyrin complexes possess phosphorescence lifetimes in the order of tens and hundreds of microseconds, respectively. Immobilization of these indicators in polymers with moderate oxygen permeability such as polystyrene and its copolymers with acrylonitrile, poly(methyl methacrylate) *etc.* results in sensors operating at ambient conditions. On the other hand, highly gas-permeable matrices are required for preparation of sensors for traces of oxygen. These include polymers with high free volume such as silicone rubber,¹³ perfluorinated polymers of Teflon AF¹⁴ and Hyflon AD¹⁵ series and poly(1-trimethylsilyl-1-propyne),¹⁶ which is the polymer with the highest oxygen permeability ever reported. In a different approach, the indicators are physically adsorbed¹⁷ or covalently¹⁴ immobilized onto the surface of porous materials such as silica-gel¹⁴ or mesoporous silica^{18,19} where the dye is fully accessible to the interaction with molecular oxygen.

Another promising way to construct in first sight “matrix-free” indicators is to incorporate the dyes into metal–organic frameworks (MOFs). MOFs are typically crystalline materials that generally consist of metal nodes and organic linkers and are usually characterized by an extremely high porosity.²⁰ The great variety of organic linkers and metal nodes enables a huge structural diversity of MOFs, including the synthesis of luminescent variants.²¹ Due to the permanent porosity and high accessibility to gases in the framework, these materials are promising for sensing applications²² including oxygen quantification.²³ The oxygen sensing unit can either be a structural element of the porous network or, alternatively, be incorporated into the pores during synthesis²⁴ or *via* post-synthetic incubation.²⁵ Incorporation into the pores during synthesis does not allow for much control over the distribution of the indicator dye within the framework that is reflected in distinctly nonlinear Stern–Volmer plots due to different oxygen accessibilities within the crystal.²⁴ Additionally, such materials may be prone to leaching of the indicator out of the pores although it can be minimized *via* design of the pore structure.²⁶ On the other hand, incorporation of a sensing element as a MOF building block not only eliminates leaching but also potential aggregation of dyes that may negatively affect the luminescent and sensing properties. Porphyrins are excellent organic linkers for the construction of MOFs and many different frameworks have been described over the last decades.^{27,28} Unfortunately, many of them are prone to hydrolysis²⁹ and are therefore of limited interest for sensing applications under ambient conditions. One well-described and hydrolytically considerably stable porphyrin-based MOF is PCN-224.³⁰ PCN-224 consists of Zr₆ clusters which are connected by tetrakis-(4-carboxyphenyl) porphyrin (H₂TCPP) as organic linker and is sufficiently insoluble in most common solvents as well as hydrolytically stable in a broad pH range (1 to 11). This MOF has a permanent porosity with a reported BET surface of 2600 m² g^{−1} and large pore sizes of 1.9 nm in

diameter.³⁰ These properties render PCN-224 and closely related MOFs attractive for many applications such as (photo)catalysis,^{31–33} photodynamic therapy^{34,35} or drug delivery.^{27,36,37} However, the luminescence properties of the PCN-224 in context of oxygen sensing in the gas phase have not been investigated. Only the Pt(II) derivative of PCN-224, Pt(II)PCN-224, was used for real-time monitoring of dissolved oxygen in aqueous dispersion. The MOF was demonstrated to respond to oxygen in physiologically relevant concentration range showing about 3-fold quenching of luminescence intensity in air-saturated conditions.³⁸ Such sensitivity is comparable to that of the materials prepared by immobilization of Pt(II) porphyrins in polystyrene or poly(methylmethacrylate) and is much lower than it would be expected from highly porous MOFs.^{39–41}

In this work we investigate the effect of molecular oxygen on luminescent properties of three MOFs: the fluorescent PCN-224 based on the metal-free porphyrin linker and the phosphorescent analogues Pt(II)PCN-224 and Pd(II)PCN-224. We will show that the porous framework ensures excellent accessibility of the porphyrin emitters to molecular oxygen and thus unprecedentedly high bimolecular quenching constants. Very efficient quenching in the gas phase is observed already for fluorescent PCN-224 MOF, whereas the luminescence of the metalated analogues is quenched by traces of molecular oxygen. Pd(II)PCN-224 MOF belongs to one of the most oxygen sensitive materials reported so far. In fact, the sensitivity of the Pd(II)PCN-224 MOF exhibiting a Stern–Volmer constant (K_{SV}) of 2610 kPa^{−1} surpasses by far the sensitivity of known trace oxygen sensors based on Pd(II) porphyrins immobilized into porous silica-gel matrix (K_{SV} of 67 kPa^{−1}).¹⁴ In this contribution we will also show how the MOFs can be conveniently immobilized onto different materials, a step necessary to improve their handling in sensing applications.

Experimental

Chemicals and materials

5,10,15,20-Tetrakis-(4-carboxyphenyl)-21,23H-porphyrin (H₂TCPP) was purchased from Porphyrin-Laboratories (Porphyrin-Laboratories GmbH, Gleschendorf, Germany, www.porphyrin-laboratories.com). Zirconyl chloride octahydrate (ZrOCl₂ × 8H₂O), *N,N*-dimethylformamide (DMF), acetic acid, potassium chloride and magnesium chloride were obtained from Roth (www.carlroth.com). Magnesium nitrate was acquired from Merck (www.merckmillipore.com). Poly(acrylonitrile) (PAN, M_w = 150 000 g mol^{−1}) and glass microfibre filters (Whatman, Ø 4.7 cm) were obtained from Sigma Aldrich (www.sigmaaldrich.com). Hyflon AD 60 was purchased from Solvay (www.solvay.com) and perfluorodecalin was from Fluorochem (www.fluorochem.co.uk). The polyamide (PA) substrate (Biodyne Plus[®] Transfer membrane with 0.45 µm pore size) was from Pall (Pall Corporation, Port Washington, USA, www.pall.com). Silica-gel coated TLC-plates and acetone were from VWR (www.vwr.com). Polyethylene terephthalate (PET) foils (125 µm) (MELINEX 506) were supplied by Pütz (Pütz GmbH Co. Folien KG, Taunusstein, Germany;



www.puetz-folien.com). Pt(II)TCPP was prepared similar to a previously described procedure⁴² and Pd(II)TCPP was prepared following a previously described microwave-assisted synthesis.⁴³ Poly(trimethylsilylpropyne) (PTMSP) was synthesized according to literature.⁴⁴

Synthesis of PCN-224. The synthesis of PCN-224 relies on a published recipe.⁴⁵ In a typical experiment H₂TCPP (6.2 mg, 7.8 μmol) and ZrOCl₂ × 8H₂O (32.2 mg, 0.1 mmol) were dispersed in a mixture of DMF (12.5 mL) and acetic acid (6.2 mL). The dispersion was sonicated for 20 min in a 50 mL round bottom flask and kept at 65 °C for 72 h. After cooling to ambient temperature, the PCN-224 crystals were collected by centrifugation and washed three times with DMF (10 mL) and three times with acetone (10 mL). For removal of residual DMF, the solid was dispersed in acetone (3 mL) and volatiles were allowed to evaporate upon heating the dispersion placed on a watch glass on a hotplate operated at 65 °C. This operation was repeated 4 times. Finally, the material was dried under vacuum at 65 °C until constant weight was reached.

Synthesis of Pt(II)PCN-224 and Pd(II)PCN-224. Pt(II)PCN-224 and Pd(II)PCN-224 were prepared analogously to PCN-224 using Pt(II)TCPP (7.7 mg, 7.8 μmol) or Pd(II)TCPP (7 mg, 7.8 μmol) as the starting materials. At the final stage, the crystals were activated for 72 h in a glass vacuum oven (B-585 from Büchi, www.buchi.com) at 0.7 mbar and 120 °C.

Immobilization of PCN-224. In order to embed PCN-224 into a Hyflon AD 60 layer, a “cocktail” containing 1% (w/w) of MOF powder and 10% (w/w) Hyflon AD 60 in perfluorodecaline was prepared. This composition was knife-coated on PET support (thickness of the wet layer 25 μm) and dried in vacuum overnight.

Immobilization experiments on glass fibres, electrospun PAN nanofibers (see ESI† for more details on preparation of electrospun nanofibers), silica-gel TLC substrate and PA filter sheets were done by directly growing the PCN-224 MOF on the substrate. The flat support materials were cut in squares of approximately 10 × 10 mm and were immersed in the solution containing MOF precursors and the synthesis was carried out as described above. After the synthesis, residual reactants were removed by soaking the materials in DMF (10 mL) and acetone (10 mL). Each solvent was exchanged three times. The materials were dried in vacuum overnight.

Instruments and methods. Powder X-ray diffraction (PXRD) measurements were performed at the XRD1 beamline at the Elettra Synchrotron in Trieste. This beamline has an operating wavelength of 1.4 Å with a beam size of 200 × 200 μm. The data was collected on a stationary Dectris Pilatus 2M detector which was mounted 400 mm away from the sample. The sample itself consisted of the MOF PCN-224 powder which was filled in a glass capillary (1.5 mm diameter and 0.02 mm wall thickness). All data were transformed to reciprocal space for analysis. All data conversion, treatment and analysis steps were performed with GIDVis.⁴⁶ In case of Pd(II)PCN-224 PXRD measurements were performed with a PANalytical Empyrean system using a sealed copper tube together with a multilayer mirror for generating a parallelized and monochromatized primary X-ray beam.

The morphology of the MOF powder samples was studied by a standard scanning electron microscope Jeol JSM-6490 LV (www.jeol.de). For preparing the samples the MOF powder was drop casted onto a Si wafer and carbon coated.

N₂ adsorption isotherms were acquired on a Micromeritics 3Flex instrument (www.micromeritics.com). Prior to the gas sorption experiment, already activated MOF samples were additionally treated at 120 °C in vacuum for 8 h. The specific surface area was calculated using the Brunauer–Emmett–Teller method.

¹H NMR spectra were recorded on a 300 MHz Avance III spectrometer from Bruker (www.bruker.com). Absorption spectra were measured on a Varian Cary 50 UV–VIS spectrophotometer from Agilent Technologies (www.agilent.com). Luminescence spectra were recorded on a Fluorolog-3 luminescence spectrometer from Horiba (www.horiba.com) equipped with a NIR sensitive photomultiplier R2658 from Hamamatsu (www.hamamatsu.com). An optical fibre setup was used to guide the excitation light to a 10 mL Schlenk flask or a 3 mL screw-cap glass vial containing the MOF sample and to guide back the emission to the monochromator and photodetector. A custom-made 3D printed adapter (ESI† Fig. S29) was used for fixation of optical fibres. Luminescence decay times were acquired on the same spectrometer equipped with the DeltaHub module. The set-up was similar to the one used in the intensity measurements but a 453 nm laser diode “NanoLED” from Horiba (www.horiba.com) was used for excitation of PCN-224 and a 405 nm “SpectraLED” from Horiba (www.horiba.com) for excitation of Pt(II)PCN-224 and Pd(II)PCN-224. DAS-6 analysis software was used for data analysis.

Photostability of fluorescent PCN-224 and phosphorescent Pt(II)PCN-224 was evaluated by continuously irradiating the samples at 420 nm or at 400 nm, respectively (Xe Lamp of Fluorolog-3 spectrometer, excitation slit 12 nm). During illumination, the PCN-224 MOF was exposed to alternating atmospheres of N₂ (10 min) and compressed air (20 min). In case of Pt(II)PCN-224 0.2% O₂ in nitrogen was used instead of compressed air. In both cases the photon flux of 48 μmol s^{−1} m^{−2} was measured with a LI2050 Light Meter from LICOR biosciences (www.licor.com) which corresponds to the irradiance of ~1.4 mW cm^{−2}. A description of the custom-made setup is available in the ESI† (Fig. S23 and S24).

The composition of the calibration gas was adjusted with help of mass-flow controllers from Voegtlin (www.voegtlin.com, Aesch, Switzerland) with the total flow rate of 200 mL min^{−1}. For calibration of PCN-224 nitrogen (99.999% purity) and compressed air were used; for Pt(II)PCN-224 nitrogen of 99.9999% purity was mixed with the test gas (0.2% O₂ in nitrogen) and for Pd(II)PCN-224 nitrogen of 99.99999% purity was mixed with a test gas containing 20 ppm oxygen in nitrogen. All test gases were acquired from Linde gas (www.linde-gas.de). The temperature was 25 °C in all the experiments. Calibrations of the two MOFs PCN-224 and Pt(II)PCN-224 under temperature controlled environments (12 °C, 17 °C, 22 °C and 35 °C) were done with a custom-made setup (Fig. S23 and S24, ESI†).



In order to investigate the effect of humidity on the oxygen response of PCN-224, the gas compositions were bubbled through saturated aqueous solutions of MgCl_2 (33% r.h.), $\text{Mg}(\text{NO}_3)_2$ (53% r.h.) and KCl (84% r.h.) before entering the measurement compartment with MOF.

Results and discussion

Synthesis and structural characterization

The PCN-224 framework is grown from a solution of H_2TCPP and zirconyl chloride octahydrate in a mixture of DMF and acetic acid.⁴⁵ The MOFs $\text{Pt}(\text{II})\text{PCN-224}$ and $\text{Pd}(\text{II})\text{PCN-224}$ were synthesized similarly using the $\text{Pt}(\text{II})$ and $\text{Pd}(\text{II})$ complexes of H_2TCPP , respectively. Residuals of DMF were removed by repeated addition and evaporation of acetone and the MOFs were activated *via* prolonged treatment in vacuum at elevated temperature (72 h, 0.7 mbar, 120 °C).

The crystal size and morphology of PCN-224, $\text{Pt}(\text{II})\text{PCN-224}$ and $\text{Pd}(\text{II})\text{PCN-224}$ was investigated using scanning electron microscopy (Fig. S15–S17, ESI† respectively) and cubical crystals with an edge length of around 3–4 μm were obtained in all cases.

Powder X-ray diffraction studies were conducted and the diffractograms were compared to those of PCN-224 and the analogous MOF based on the corresponding $\text{Ni}(\text{II})$ complex of TCPP ($\text{Ni}(\text{II})\text{TCPP}$).³⁰ In both of these reference cases,

the structure is cubic and belongs to the space group $Im\bar{3}m$ (no. 229). Half of the edges of the Zr_6 -clusters forming the nodes are bridged by carboxylates from the TCPP ligands and the TCPP linker vacancies appearing in an ordered 3D checker-board fashion (Fig. 1). To mimic the $\text{Pt}(\text{II})$ metalated MOF prepared here, the $\text{Ni}(\text{II})$ in the literature structure was changed for $\text{Pt}(\text{II})$. Fig. 1 shows that for both model systems $\text{Ni}(\text{II})\text{PCN-224}$ and such a hypothetical $\text{Pt}(\text{II})\text{PCN-224}$ the relative intensities of all peaks agree well with the relative intensities determined in the experiment. By comparing peak positions and relative intensities between the measured data and the data calculated for the literature structure (Fig. 1b for $\text{Pt}(\text{II})\text{PCN-224}$ and ESI† Fig. S4) it is concluded that the synthesized MOF exhibits the same structure as the PCN-224 family reported by Feng *et al.*³⁰ It is to note, that the published structures of PCN-221 (disordered variant of PCN-224) and MOF-525 (all edges of the Zr_6 cluster linked with TCPP derivatives) exhibit very similar patterns to that of PCN-224 at scattering angles below 15°. However, only PCN-224 features the superstructure reflections observed at scattering angles of 3.2 and 5.5° so that the occurrence of ordered vacancies in the crystals discussed here is prevailing.⁴⁷ Powder X-ray diffraction data obtained for $\text{Pd}(\text{II})\text{PCN-224}$ are very similar to the those obtained for PCN-224 and its $\text{Pt}(\text{II})$ analogue (Fig. S10 and S11, ESI†).

Calculated BET surface areas of the three MOFs PCN-224, $\text{Pt}(\text{II})\text{PCN-224}$ and $\text{Pd}(\text{II})\text{PCN-224}$ obtained from N_2 sorption

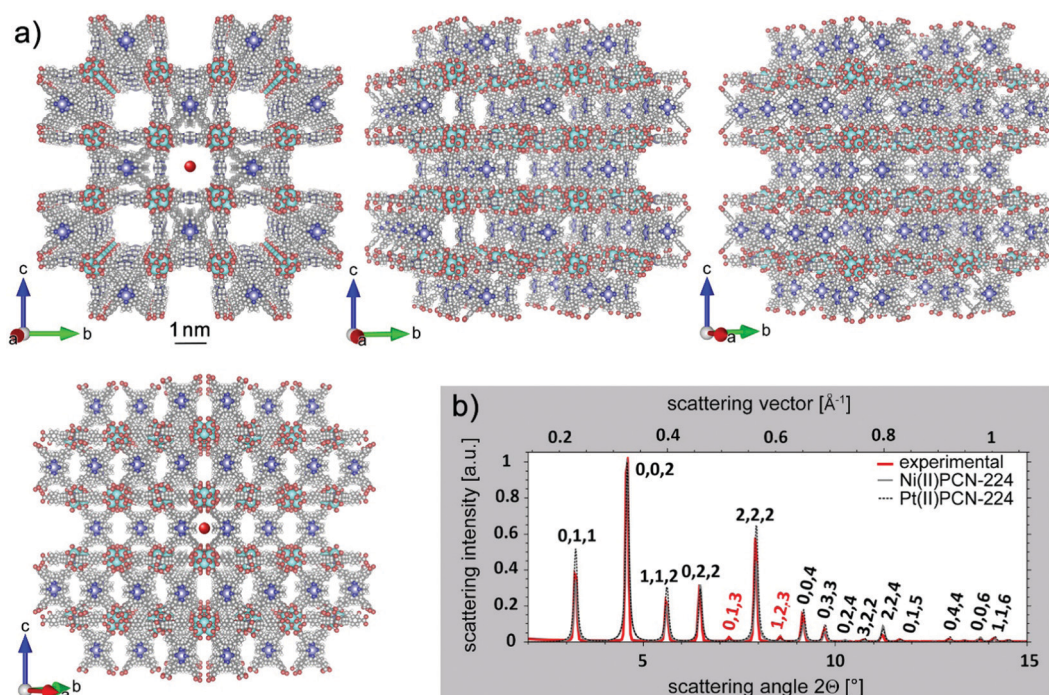


Fig. 1 (a) Space fill-model of a metalated PCN-224 drawn in perspective view along the *a*-axis and rotated for 15, 30 and 45° around the *c*-axis showing the channels through the MOF and a red sphere representing the size of O_2 (based on the kinetic diameter of 346 pm) shown for comparison. The visualization was done with the aid of VESTA.⁴⁸ (b) Normalized PXRD data shown as a line graph. Experimental and calculated data for $\text{Pt}(\text{II})\text{PCN-224}$ are plotted together with literature data for $\text{Ni}(\text{II})\text{PCN-224}$. The data was extracted by integrating over the intensity of the experimental PXRD pattern and normalized to the maximum intensity. The data for the model structures was calculated by using the Mercury software.



experiments (Fig. S12–S14, ESI†) are in very good agreement with data from literature.³⁰ PCN-224 shows the highest BET surface area of $2600 \text{ m}^2 \text{ g}^{-1}$, the BET surface areas of Pt(II)PCN-224 and Pd(II)PCN-224 are only slightly lower ($2300 \text{ m}^2 \text{ g}^{-1}$ and $2400 \text{ m}^2 \text{ g}^{-1}$, respectively). Remarkably, the herein prepared Pt(II)PCN-224 has significantly higher surface area than the same material prepared *via* a different procedure³⁸ ($2300 \text{ m}^2 \text{ g}^{-1}$ vs. $1700 \text{ m}^2 \text{ g}^{-1}$ respectively). The high surface areas of the PCN-based MOFs underline very high permanent porosity of the materials and excellent accessibility of the luminescent indicator centres (H_2TCPP , Pt(II)TCPP and Pd(II)TCPP) in the framework.

Luminescence of PCN-224 MOFs

Emission spectra of the three MOFs PCN-224, Pt(II)PCN-224 and Pd(II)PCN-224 are shown in Fig. 2a. In case of PCN-224 the emission is attributed to fluorescence due to decay time in the nanosecond time domain ($\tau = 6.7 \text{ ns}$ in the absence of O_2). In contrast, the MOFs based on Pt(II) and Pd(II) porphyrins show phosphorescence ($\tau = 18.7$ and $390 \mu\text{s}$, respectively). As can be seen, the emission maximum for the PCN-224 in free powder is located at significantly longer wavelength (711 nm) than expected and only a small shoulder at around

660 nm is visible. Such distortion can be attributed to the inner-filter effect that is particularly strong in the MOF due to high concentration of the porphyrin and strong overlap between fluorescence and the Q_x (0,0) absorption band. The same phenomenon is well known for concentrated solutions of fluorescent emitters including porphyrins.⁴⁹ In accordance to the properties of Pt(II) and Pd(II) porphyrins in solutions, the emission maxima of Pt(II)PCN-224 is located at shorter wavelength ($\lambda_{\text{max}} = 645 \text{ nm}$) compared to the corresponding Pd(II) MOF ($\lambda_{\text{max}} = 675 \text{ nm}$).⁵⁰ The structure of PCN-224-based MOFs already suggests that individual porphyrin molecules are isolated and their aggregation, which causes pronounced changes of the optical properties⁵⁰ of the materials, is mitigated. Indeed, the excitation spectra recorded for Pt(II)PCN-224 (in aqueous dispersion) and the respective building block Pt(II)TCPP (in aqueous solution) are very similar (Fig. 2b). Thus, the arrangement of porphyrin molecules in MOF appears to be an efficient strategy to obtain systems with high concentration of the luminophore without its aggregation.

Luminescence quenching by molecular oxygen

The luminescence of the three MOFs was found to be quenched by molecular oxygen (Fig. 2c). While such behaviour was

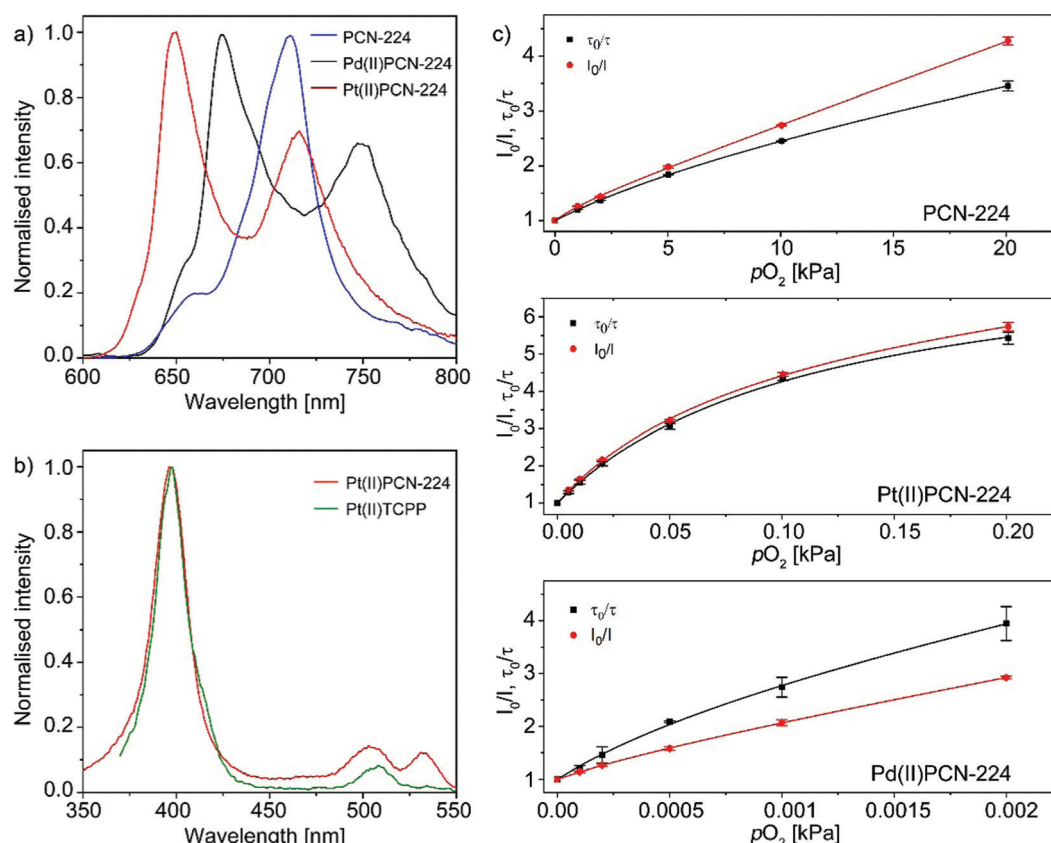


Fig. 2 Luminescence spectra of MOFs; (a) emission spectra of PCN-224 (λ_{exc} 420 nm, blue line), Pt(II)PCN-224 (λ_{exc} 400 nm, red line) and Pd(II)PCN-224 (λ_{exc} 400 nm, black line) powders under anoxic conditions; (b) excitation spectra of Pt(II)TCPP in aqueous solution and Pt(II)PCN-224 in aqueous dispersion (λ_{em} 650 nm) under anoxic conditions. (c) Stern–Volmer plots for PCN-224, Pt(II)PCN-224 and Pd(II)PCN-224 at 25 °C and dry gas atmosphere.

expected for the MOFs based on phosphorescent Pd(II) and Pt(II) porphyrins it is rather unusual in the case of PCN-224 formed by a fluorescent free-base porphyrin. In fact, fluorescent dyes are usually seen as “immune” to oxygen quenching due to their short emission lifetimes. Luminescence quenching by molecular oxygen is of dynamic nature and the Stern–Volmer eqn (1) is used to adequately describe the quenching dependency in case of dissolved dyes. In these conditions, the decrease of the luminescence intensity is identical to the decrease of the decay time and both show linear dependency on the concentration of the quencher [O₂].

$$\frac{I_0}{I} = \frac{\tau_0}{\tau} = 1 + K_{SV} \cdot [O_2] = 1 + k_q \cdot \tau_0 \cdot [O_2] \quad (1)$$

where I_0 and τ_0 are the luminescence intensity and luminescence decay time under oxygen-free conditions, I and τ the same parameters at a certain oxygen concentration, K_{SV} is the Stern–Volmer constant and k_q the bimolecular quenching constant. For indicator dyes immobilized into different matrices (most commonly polymers) such ideal dependency is rarely found. Instead, the dependencies are not linear and not fully identical for the luminescence intensity and the decay time. This is explained by heterogeneity of the environment in which individual indicator molecules are located that results in different oxygen quenchability. As can be seen from Fig. 2c, the same is true for the three MOFs. Such behaviour can be due to combination of several factors including (i) defects in the MOF, (ii) different oxygen accessibility of the porphyrin molecules that are located on the surface of an individual crystal and on the inner pores, (iii) blocking of some pore channels for instance with residual solvent molecules that are trapped inside and hinder oxygen transport. Therefore, the so called “two site model”⁵¹ (eqn (2)) was used to fit the experimental data. The model postulates localization of the indicator in two environments of significantly different oxygen permeability and although is physically meaningful only for the luminescence intensity, it also adequately fits the decay time plots.

$$\frac{I_0}{I} = \left(\frac{f}{1 + K_{SV1} \cdot [O_2]} + \frac{f-1}{1 + K_{SV2} \cdot [O_2]} \right)^{-1} \quad (2)$$

where f is the proportion of the dye localized in the first environment, K_{SV1} and K_{SV2} are the Stern–Volmer constants for the first and the second environment. Fig. 2c shows that already for PCN-224 based on free-base porphyrin the quenching is very efficient: the fluorescence intensity decreases by more than 4-fold at air saturation comparing to anoxic conditions. For comparison, the quenching of the same magnitude is observed for Pt(II) porphyrins (such as Pt(II)TFPP) immobilized in polystyrene.⁵² Notably, the luminescence decay time of Pt(II)TFPP in polystyrene⁵² is roughly 4 orders of magnitude longer (55 μ s) than that of H₂TCPP in PCN-224 (6.7 ns). Such an efficient quenching in a MOF can be explained by the fact that the porphyrin molecules are structural elements of the porous channels and thus are freely exposed to oxygen molecules that diffuse through them. In contrast, even in polymers with high free-volume, polymer chains surround the indicator molecule partly protecting

it from interaction with oxygen. It is therefore likely that oxygen permeability in PCN-224 is higher than that of poly(trimethylsilylpropyne) (PTMSP), a polymer with the highest oxygen permeability known.¹ To verify this, we embedded a lipophilic analogue of H₂TCPP, namely 5,10,15,20-tetrakis-(4-methoxycarbonylphenyl)-21,23H-porphyrin (H₂TMCP) into PTMSP and investigated its oxygen response (Fig. S27, ESI†). The Stern–Volmer constant obtained for this material is 0.073 kPa^{−1} which is about 3.5-fold lower than for PCN-224 (0.25 kPa^{−1}) despite the longer τ_0 of H₂TMCP in PTMSP (10.7 ns).

A known strategy to obtain highly oxygen-sensitive materials is to immobilize the indicator on the surface of a porous material such as silica-gel¹⁴ or porous aluminium oxide.⁵³ In such materials the indicator molecules are exposed to the gas that more or less freely diffuses through the channels. As can be seen, the fluorescence of H₂TCPP adsorbed on the surface of porous silica-gel is indeed significantly quenched by molecular oxygen (Fig. S28, ESI†). The Stern–Volmer constant of 0.19 kPa^{−1} is however lower than that for PCN-224 despite the fluorescence decay time being higher for H₂TCPP (τ_0 of ~10 ns) than for the porphyrin in the MOF (τ_0 = 6.7 ns).

Considering the significant quenchability of PCN-224 that relies on a fluorescent dye, we expected much stronger sensitivity for phosphorescent Pt(II)PCN-224 and Pd(II)PCN-224. Experimental results (Fig. 2c) show that this is indeed the case. In fact, in case of the former more than 5-fold decrease of luminescence intensity is observed at 200 Pa pO₂, whereas Pd(II)PCN-224 shows 3-fold quenching at only 2 Pa of oxygen. The limits of detection (s) estimated with the blank value method (blank value + 3 × standard deviation) were found to be 1 and 0.015 Pa for Pt(II)PCN-224 and Pd(II)PCN-224, respectively. The materials based on Pt(II) and Pd(II) porphyrins thus cover trace and ultra-trace range of oxygen concentrations. Since the Stern–Volmer constant K_{SV} is proportional to the luminescence decay time of the indicator dye (eqn (1)), it is interesting to compare the bimolecular quenching constant $k_q = K_{SV}/\tau_0$ that reflects how efficient the diffusion of oxygen in the porous material is. The bimolecular quenching constant k_q was calculated using the K_{SV} values obtained from the luminescence intensity plots. The k_q constants are 37 000, 3900, and 6700 Pa^{−1} s^{−1} for PCN-224, Pt(II)PCN-224 and Pd(II)PCN-224, respectively (Table 1). The bimolecular quenching constant is about one order of magnitude higher for the MOF based on fluorescent free-base porphyrin compared to the phosphorescent MOFs. This is in good agreement with the expectations of k_q approaching diffusion controlled limit (k_{diff}) for fluorescent dyes (quenching of the excited singlet state) and 1/9 k_{diff} for phosphorescent dyes (quenching of the excited triplet state) due to contribution of the spin-statistical factor.⁵⁴ For comparison, the k_q constant for H₂TMCP embedded into PTMSP was calculated to be 7200 Pa^{−1} s^{−1}, that is about 5-fold lower than that for PCN-224 indicating that oxygen permeability of the MOF is significantly higher than that of PTMSP. Although the arrangement of the porphyrin molecules in PCN-224 may be particularly advantageous for efficient oxygen quenching, such property is not truly unique and may be characteristic for many



Table 1 Oxygen sensing properties of selected luminescent porous and non-porous materials

Material	Matrix	τ_0	K_{SV} [kPa ⁻¹]	k_q [Pa ⁻¹ s ⁻¹]	LOD [Pa]	I_0/I^e	Ref.
SUMOF-6-Eu	Porous	1 ms	0.07	0.07	450	I_0/I_{101} kPa ~ 7.66	57
EuNDC-MOF	Porous	1.33 ms	0.13	0.1	21	I_0/I_{101} kPa ~ 15.4	58
Pt(II)Porph:UiO	Porous	28 μ s	0.13	4.7	n.a.	τ_0/τ_{20} kPa ~ 4	59
Pd(II)OEP in PTMSP	Polymer	~ 770 μ s	17	$\sim 22^d$	300	I_0/I_6 kPa ~ 100	60
Pt(II)BP in silicone rubber	Polymer	50 μ s	1.4	28	n.a.	τ_0/τ_2 kPa ~ 4	13
Pt(II)TFPP in Hyflon AD60	Polymer	92 μ s	3.9	42	4^f	τ_0/τ_2 kPa $\sim 5^f$	61
Pd(II)TFPP in Hyflon AD60	Polymer	1180 μ s	62.4	53	0.4^f	τ_0/τ_{50} kPa $\sim 3^f$	61
Pt(II)TFPP on silica-gel particles	Porous	70 μ s	3.7	53	0.27	$\tau_0/\tau_{0.2}$ kPa ~ 2	14
Pt(II)OEP in PTMSP	Polymer	~ 100 μ s	5.6	$\sim 56^d$	300	I_0/I_{101} kPa ~ 225	16
Pd(II)TFPP on silica-gel particles	Porous	987 μ s	67	68	0.015	$\tau_0/\tau_{0.1}$ kPa ~ 7	14
Ru(II):MAF-34	Porous	1.18 μ s	0.21	178	n.a.	I_0/I_{101} Pa ~ 8	24
Pyrene:MAF-4	Porous	59 ns	0.06	1000	n.a.	I_0/I_{101} kPa ~ 7	62
Pt(II)PCN-224	Porous	18.6 μ s	73^a	3900	1	$I_0/I_{0.2}$ kPa ~ 5	This work
Pd(II)PCN-224	Porous	390 μ s ^b	2610^a	6700^b	0.015	I_0/I_2 Pa ~ 4	This work
H ₂ TMCP in PTMSP	Polymer	10 ns	0.072	7200	920	I_0/I_{20} kPa ~ 1.5	This work
MAF-X11	Porous	14 ns	0.27	19 000	n.a.	I_0/I_{101} kPa ~ 30	55
MAF-2	Porous	116 μ s	3.56	30	4.7	I_0/I_{101} kPa ~ 356	63
M66:MAF-2	Porous	101 μ s	1480	15 000	6.8×10^{-3}	I_0/I_4 Pa ~ 5	56
M76:MAF-2	Porous	108.5 μ s	1200	11 000	8.4×10^{-3}	I_0/I_4 Pa ~ 5	56
H ₂ TCPP on silica-gel particles	Porous	10 ns	0.19	19 000	160	I_0/I_{20} kPa ~ 2	This work
H ₂ TCPP derivative on Al ₂ O ₃	Porous	~ 10 ns	0.23	$\sim 23\,000^c$	n.a.	I_0/I_{101} kPa ~ 9	53
PCN-224	Porous	6.7 ns	0.25^a	37 000	40	I_0/I_{20} kPa ~ 4	This work

^a Ksv₁ fit according to eqn (2). For other fit parameters see Table S5 (ESI). ^b Average quenching constant based on bi-exponential decay of two luminescent species. ^c Calculated based on the estimated decay time (10 ns) reported in ref. 53. ^d Calculated based on the estimated decay times (100 μ s for Pt(II)OEP and 770 μ s for Pd(II)OEP) given in ref. 1. ^e Expressed for the respective measurement range. ^f Data from liquid phase oxygen sensing.

other luminescent MOFs. In fact, k_q constants of material such as MAF-X11⁵⁵ (19 000 Pa⁻¹ s⁻¹), M66:MAF-2 (15 000 Pa⁻¹ s⁻¹) or M76:MAF-2 (11 000 Pa⁻¹ s⁻¹)⁵⁶ are very high, but still about

2-fold lower than for PCN-224. On the other hand, the MAF-2 based materials in fact show the highest k_q values reported for phosphorescent compounds.

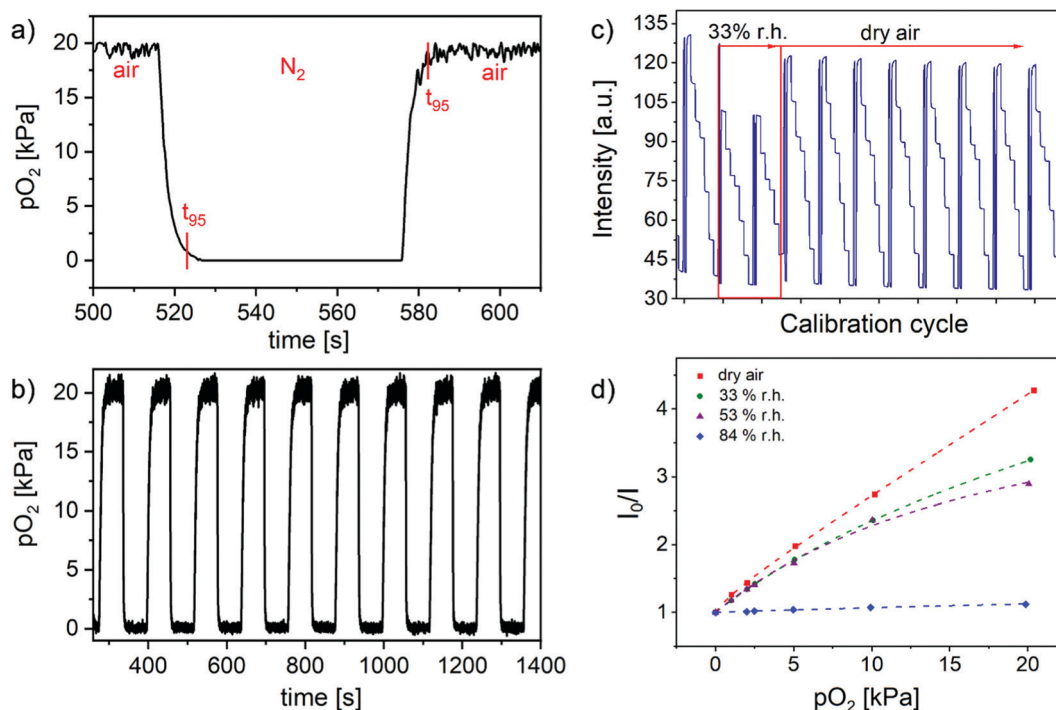


Fig. 3 Oxygen-sensing properties of free-standing PCN-224 crystals; (a) dynamic response to alteration of gas atmosphere between nitrogen and air; (b) 10 consecutive cycles of changing atmosphere between nitrogen and air demonstrating excellent repeatability and reversibility of the response; (c) dynamic response of the material in dry air (first and last eight cycles) and at 33% r.h. (cycles 2 and 3); (d) Stern–Volmer plots for PCN-224 at varying relative humidity at 25 °C.



Table 1 summarizes the oxygen-sensing properties of various reported luminescent MOFs as well as selected porous and non-porous materials based on porphyrin indicators embedded in highly oxygen-permeable matrices. It should be noted that the selection of materials was limited to those which allowed for the calculation of quenching constant k_q , *i.e.* contain the data on the luminescence decay time in the absence of the quencher. As can be seen, the bimolecular quenching constant k_q for most materials is much lower than that obtained for PCN-224, with the exception of MAF-X11,⁵⁵ MAF-2 based MOFs⁵⁶ and H₂TCPP adsorbed on other porous materials such as silica-gel and aluminium oxide. Comparison of the trace sensors based on Pt(II) and Pd(II) porphyrins reveals that these indicators immobilized in porous matrices (silica-gel) and highly oxygen-permeable non-porous polymers (Hyflon AD 60, silicone rubber, PTMSP) show the k_q values in the range of 20–70 Pa^{−1} s^{−1} that is roughly 2 orders of magnitude lower than the bimolecular quenching constant obtained for Pt(II)PCN-224 and Pd(II)PCN-224. PCN-224 and its metalated analogues thus appear the most oxygen-permeable materials ever reported.

Response times and reversibility

In a next step, response times and the reversibility were assessed. As shown in Fig. 3 the PCN-224 crystals fully reversibly respond to the alteration between oxygen-free atmosphere and air. The same holds for the two metalated MOF crystals. The dynamic response times t_{95} are within several seconds (~ 6 s going from anoxic conditions to air and ~ 7 s in the reverse direction). Although such response times are comparable to most state-of-the-art oxygen sensors based on polymer-immobilized indicators, they appear to be much longer than expected from extreme oxygen permeability of the MOF. As will be discussed below these values are likely to be greatly overestimated due to the time needed for the gas atmosphere to be exchanged in the chamber containing the crystals.

Photostability of MOFs

Continuous irradiation of PCN-224 and Pt(II)PCN-224 MOFs in timeframe of several hours was found to result in noticeable ($\sim 10\%$) loss of luminescence intensity (Fig. S30 and S31, ESI†). Photodegradation may be surprising at the first glance since Zr-based MOFs are often highlighted as materials with excellent photostabilities and stability against singlet oxygen which is generated during quenching of fluorescent and phosphorescent emitters. Numerous applications of PCN-224 which require efficient singlet oxygen production, such as photodynamic therapy,³⁴ photocatalytic degradation⁶⁴ or photodynamic sterilization⁶⁵ have been demonstrated. Photostability of PCN-224 MOF in liquid suspension was found to be excellent.³⁷ However, it should be considered that the lifetime of singlet oxygen in water is many orders of magnitude shorter than in the gas phase ($\sim 3 \mu\text{s}$ ⁶⁶ and ~ 3 s,⁶⁷ respectively) so that noticeable photodegradation in gas phase in fact can be expected. Since in a typical sensing application continuous irradiation is not required and discrete measurement points under short light exposure (10–20 ms) are acquired, moderate

photostability can be well tolerable. Importantly, experiments with closely related MOF-525, that also relies on H₂TCPP as a building block, indicated no structural change during photodegradation.⁶⁸ Since the framework itself stays intact, the limited photostability may be addressed by rendering the porphyrin moiety more photostable, for example by introducing electron withdrawing substituents in *beta* or *meso* positions of the porphyrin macrocycle.^{69,70}

Influence of temperature on sensitivity of MOFs

As in case of any chemosensor, temperature affects the response of luminescent oxygen sensors. Whereas a wealth of information is available on the temperature cross-talk of conventional oxygen sensing materials that make use of indicator dyes immobilized into polymeric or inorganic matrices, little is known on the effect of temperature on the oxygen sensitivity of luminescent MOFs. To address this, the oxygen sensing properties of the fluorescent PCN-224 and phosphorescent Pt(II)PCN-224 MOFs were investigated at 12 °C, 17 °C, 22 °C and 35 °C using a custom-made set-up (Fig. S23 and S24, ESI†). In both cases, a decrease of oxygen sensitivity with temperature was observed (Fig. S25, S26 and Tables S3, S4, ESI†). For PCN-224 the Stern–Volmer constants K_{SV1} decreases from 0.32 kPa^{−1} at 12 °C to 0.21 kPa^{−1} at 35 °C. Analogously, in case of Pt(II)PCN-224 the Stern–Volmer constant decreases from 90 kPa^{−1} to 55 kPa^{−1} for the same temperatures. Notably, within the experimental error the luminescent decay times did not change in this temperature range (Tables S3 and S4, ESI†) so that the decrease of the Stern–Volmer constant corresponds to the same decrease of the bimolecular quenching constant k_q . Since elevated temperature favours faster oxygen diffusion, the reduced quenching must be due to desorption of surface-adsorbed oxygen at higher temperatures. Experimental data regarding oxygen adsorption of several MOFs (HKUST-1, NU-125, UiO-66, N-AC) show similar temperature dependency of oxygen adsorption properties.⁷¹ The data emphasize the important role of adsorption and desorption processes in oxygen quenching in MOFs. In this respect, MOFs appear to be similar to materials prepared by immobilization of oxygen indicators on solid surfaces like silica-gel. For this group of materials significant decrease of the quenching constant at elevated temperatures has been documented^{72,73} and diffusional quenching by surface-adsorbed oxygen is assumed to be the dominating quenching pathway.⁷⁴ It should be noted that when the pores of the Pt(II)PCN-224 are completely filled with water the Stern–Volmer constant increases with temperature³⁸ due to faster oxygen diffusion and absence of other contributing factors.

Influence of humidity on oxygen sensitivity

As was shown in Fig. 2c, Pt(II)PCN-224 is extremely sensitive to molecular oxygen in gas phase showing almost 6-fold quenching of phosphorescence intensity already at 0.2 kPa O₂. On the other hand, the same material was retested to show about 3-fold quenching of luminescence intensity in air-saturated aqueous dispersion (~ 21 kPa O₂).³⁸ Such drastic difference in the



sensitivity (about 200-fold) can be explained by substitution of the gas phase inside the pores of the MOF by aqueous phase. Even for measurement in the gas phase, residual solvents including water can be trapped inside the MOF and thus hinder diffusion of oxygen through the pores. In fact, the extreme oxygen quenching capabilities of MOFs could only be achieved when the materials were fully activated by treatment in vacuum and at elevated temperature (0.7 mbar, 120 °C, 72 h). Fig. 3c and d shows the humidity influence on the sensing properties of PCN-224. As can be seen, an increase of relative humidity from 0 to 33% results in some decrease of fluorescence intensity and sensitivity of the material. The behaviour is reversible and the parameters are recovered when dry conditions are established again (Fig. 3c). Increasing relative humidity to 53% results in further decrease of sensitivity (Fig. 3d), again reversible. However, if the humidity is raised to 84%, a drastic decrease of the sensitivity is observed. Comparison of K_{SV} values at 0 and 84% relative humidity (0.25 kPa^{-1} and 0.008 kPa^{-1} , respectively) reveals the decrease of sensitivity by about 30-fold. Notably, this effect cannot be reversed by changing back to dry atmosphere but only by activation of the material at elevated temperature and vacuum. The drastic decrease of sensitivity at high humidity may be due to condensation of water inside of the porous channels of the MOF. In this respect substitution of *meso*-tetracarboxyphenyl porphyrin with (partly) fluorinated analogues may be a promising way to reduce or even completely overcome the above limitation. Porphyrins bearing fluorine atoms in β -position are known,⁷⁰ and partial fluorination in the *meso*-phenyl positions is likely to be feasible. As mentioned above such substitution is also expected to positively affect the photostability of the sensors.

Immobilization of MOFs

Potential application of MOFs for optical oxygen sensing requires their immobilization into/onto solid supports to obtain a sensing element with acceptable mechanical stability. A notable exception is the application of MOF (micro)crystals in form of aqueous dispersion.³⁸ For all the experiments, PCN-224

was chosen as the model system but results are expected to be applicable to the metalated congeners due to their high structural similarity. A straightforward way to prepare sensing materials is to disperse crystals in a polymeric matrix. To do so, the free crystals (Fig. 4a) were first dispersed in a polymer solution in an organic solvent that was evaporated after film casting. Hyflon AD 60 was chosen for its high oxygen permeability and chemical inertness.⁷⁵ Unfortunately, a drastic loss of sensitivity was observed after immobilization (Fig. 4f).

This may be due to clogging of the pores by flexible polymer chains or entrapped solvent (perfluorodecaline) that hinders oxygen from entering the MOF freely. Notably, perfluorodecaline is much less volatile than water (b.p. 142 °C) and is much larger so it may be very difficult to remove it once it gets entrapped in the pores of the MOF. Moreover, removal of residual solvent from the perfluorinated polymer itself was reported to be challenging.⁷⁶

As an alternative approach, the growth of the MOF on different substrates was investigated.⁷⁷ PCN-224 was successfully grown on heat-stabilized poly(acrylonitrile) (PAN) nanofibers (Fig. 4b). PAN nanofibers were prepared *via* an electrospinning technique and a cross-linking step at high temperature (280 °C) that renders the fibres insoluble in DMF.⁷⁸ This property enables the MOF to grow from a solution of the precursors in DMF. SEM revealed MOF crystals grown not only on the surface of the fibres but also around the fibres (Fig. S21, ESI†) and PXRD confirmed the structure of PCN-224 grown on the fibrous support (Fig. S22, ESI†). The oxygen sensitivity of the resulting material is very similar to that of the individual crystals (Fig. 4f). This is not the case for several other composite materials. MOF crystals grown on different Si-based materials such as a glass fibre filter (Fig. 4c and Fig. S18, ESI†) and aluminium-supported silica-gel (Fig. 4d and Fig. S20, ESI†) exhibited some decrease of oxygen sensitivity (Fig. 4f). The decrease in sensitivity may be due to blocking of some part of the crystals by the virtually oxygen-impermeable glass/silica-gel backbone. PCN-224 was also grown on a flat polyamide (PA) filter (Fig. 4e and Fig. S19, ESI†) and a similar decrease in the sensitivity was observed (Fig. 4f). For the flat

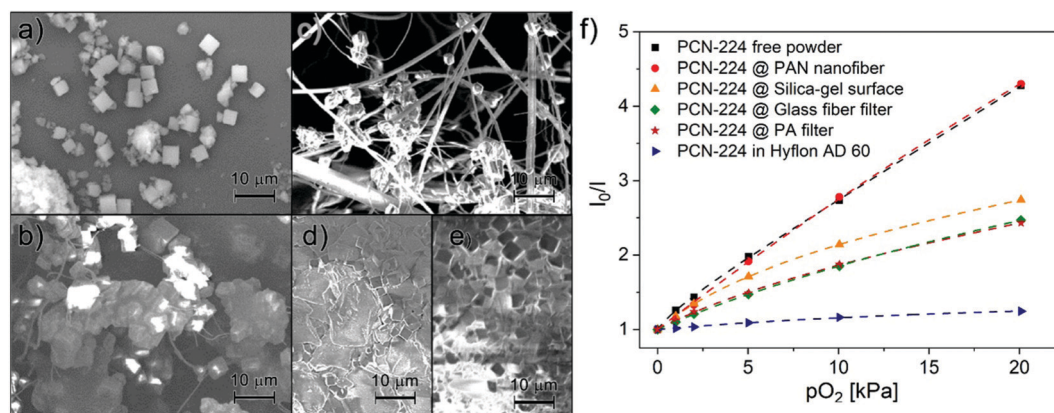


Fig. 4 Left: SEM images of the PCN-224 MOF; (a) free MOF powder; (b) immobilized on electrospun poly(acrylonitrile) nanofibers; (c) immobilized on glass fibres of a glass fibre filter; (d) crystals grown on silica-gel; (e) grown on flat PA filter surface. Right: (f) Stern–Volmer plots for PCN-224 immobilized onto/in various matrices.



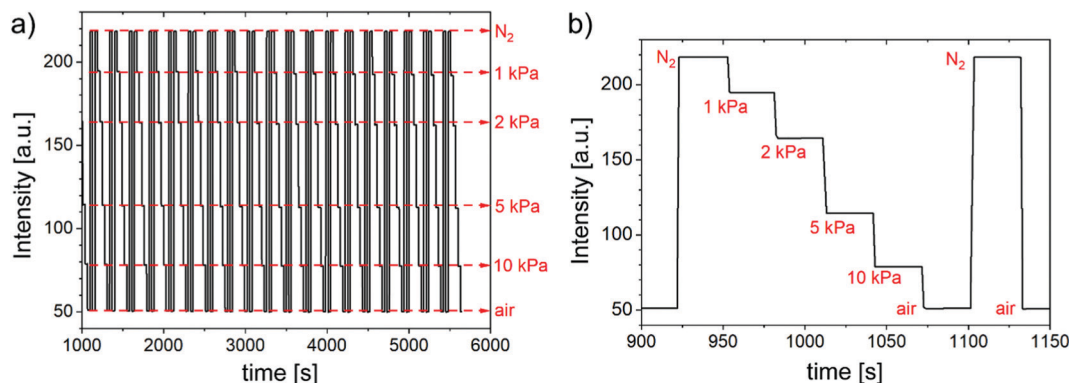


Fig. 5 (a) Fluorescence intensity response of PCN-224 grown on PAN nanofibers to alteration of oxygen content in the gas atmosphere during 19 consecutive calibration cycles; (b) dynamic response of PCN-224 grown on PAN nanofibers.

silica-gel and PA supports the individual crystals are comparable in size and the oxygen sensitivity is also very similar.

Finally, the sensing properties of PCN-224 grown on PAN nanofibers were investigated in more detail. The response of the material is very fast and fully reversible (Fig. 5a) and is generally similar to that of the free crystals. It should be noted that since the compact PCN-224 crystals are grown around the individual PAN fibres, their position in the sensor material is fixed and there is no possibility of movement of individual crystals anymore. This allows for a very fast exchange of the gas atmosphere so that the measured response times are closer to the true values (in particular when compared to the measurement performed with the free crystals presented in Fig. 4). The full response (t_{100}) in both directions was accomplished within 1 s (Fig. 5b), which again appears to reflect the time needed for the gas exchange. The true response times in terms of more commonly used t_{95} (time needed for 95% of the signal change) are expected to be well below 1 s. During 19 consecutive calibration cycles, almost no drift (less than 2% over a continuous illumination of more than one hour and 19 consecutive cycles) was found and no change in sensing behaviour was observed. Therefore, immobilization by growing PCN-224-related materials on PAN nanofibers can be considered highly promising for preparation of solid state (sensing) materials.

Conclusions

Porous MOFs prepared from luminescent porphyrins as building blocks show extremely interesting oxygen sensing properties in the gas phase. Compared to conventional oxygen-sensing materials it becomes possible to suppress aggregation of dye molecules and achieve high indicator content in the sensing material. High porosity and “free-standing” dye molecules are responsible for the unprecedentedly efficient quenching of the luminescence by molecular oxygen. Significant quenching is observed already for the MOF based on a fluorescent porphyrin dye despite its short fluorescence decay time of several nanoseconds. Phosphorescence of MOFs that make use of analogous Pt(II) and Pd(II) porphyrins is quenched by trace amounts of oxygen. The bimolecular quenching

constant k_q is extremely high and exceeds by several folds the value obtained for the porphyrins immobilized in poly(trimethylsilylpropyne), the polymer with the highest oxygen permeability known. MOF growth on porous support materials paves the way for their practical applications in oxygen sensing in gas phase. Among several materials investigated, cross-linked poly(acrylonitrile) nanofibers were found to be particularly promising, since the oxygen-sensing characteristics are virtually the same as for the free crystals. The material shows fast and fully reversible response. A current limitation of the PCN-224 based MOFs in oxygen sensing is the cross-talk to humidity and the drastic decrease of the sensitivity at high humidity. Substitution of the dye molecules used as building blocks by fluorinated porphyrins may help to overcome the current limitation.

Author contributions

T. B. performed most experiments, analysed the data and prepared the initial draft of the manuscript. C. W. performed the structural characterization of the metal-organic frameworks. I. D. performed some of the experiments. C. S. and S. B. conceived, supervised and directed the study. All authors contributed to the analysis of the result and to the writing of the manuscript.

Conflicts of interest

The authors have no conflict of interest to declare.

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