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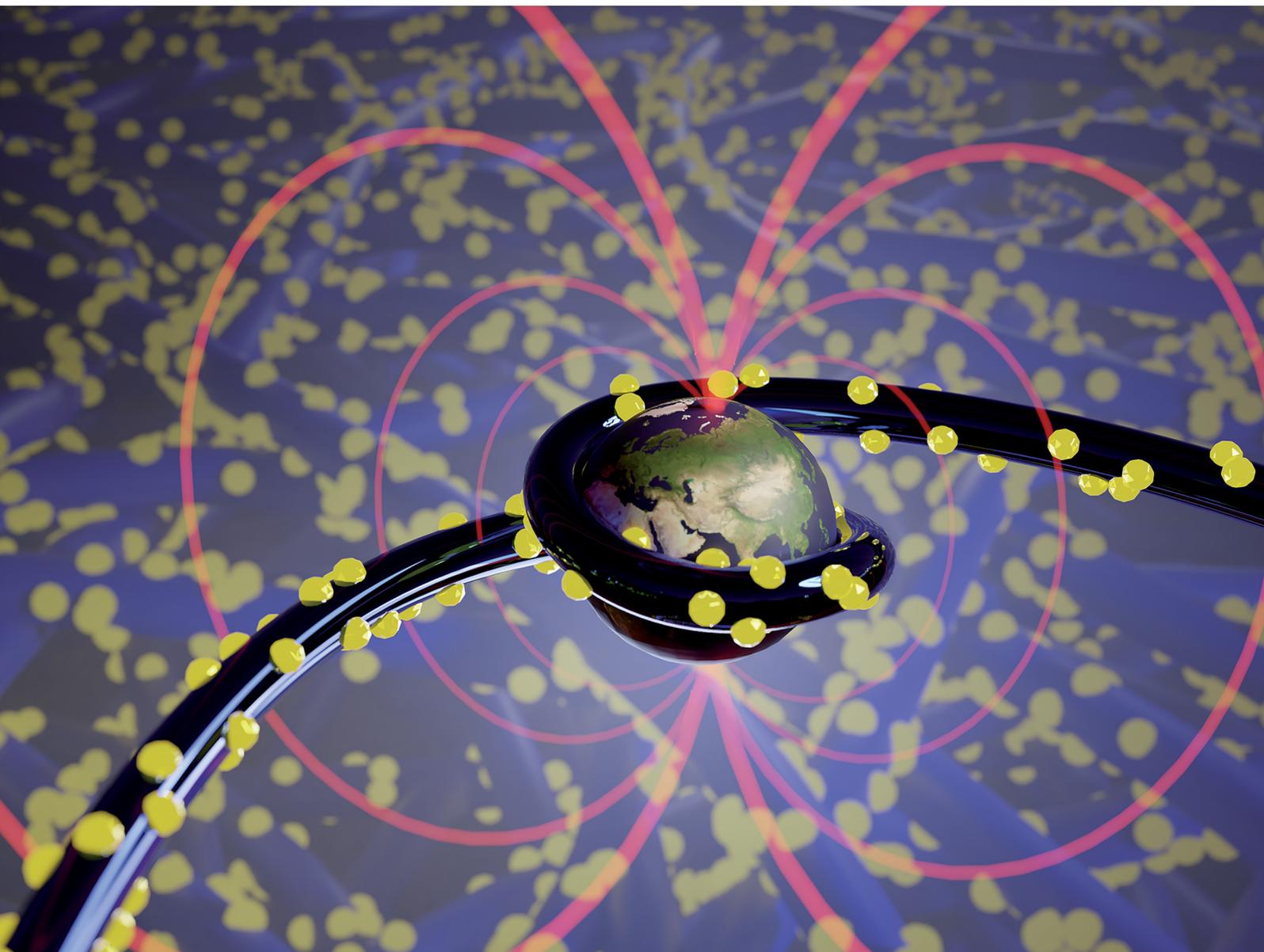
REVIEW ARTICLE
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A review on electrospun magnetic nanomaterials:
methods, properties and applications

REVIEW

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A review on electrospun magnetic nanomaterials: methods, properties and applications

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Magnetic materials display attractive properties for a wide range of applications. More recently, interest has turned to significantly enhancing their behaviour for advanced technologies, by exploiting the remarkable advantages that nanoscale materials offer over their bulk counterparts. Electrospinning is a high-throughput method that can continuously produce nanoscale fibres, providing a versatile way to prepare novel magnetic nanomaterials. This article reviews 20 years of magnetic nanomaterials fabricated *via* electrospinning and introduces their two primary production methods: electrospinning polymer-based magnetic fibres directly from solution and electrospinning fibrous templates for post-treatment. Continual advances in electrospinning have enabled access to a variety of morphologies, which has led to magnetic materials having desirable flexibility, anisotropy and high specific surface area. Post-treatment methods, such as surface deposition, carbonization and calcination, further improve or even create unique magnetic properties in the materials. This renders them useful in broad ranging applications, including electromagnetic interference shielding (EMS), magnetic separation, tissue engineering scaffolding, hyperthermia treatment, drug delivery, nanogenerators and data storage. The processing methods of electrospun magnetic nanofibres, their properties and related applications are discussed throughout this review. Key areas for future research have been highlighted with the aim of stimulating advances in the development of electrospun magnetic nanomaterials for a wide range of applications.

1. Introduction

Materials that possess magnetic character often exhibit specific desirable properties. This has led to the inclusion of magnetic materials in an ever-growing range of applications, including hard magnetic materials within magneto-biology,[1] magnetic medicine,[2] magnetic separation[3] and electrical machinery and soft magnetic materials in stator or rotator parts of generators and motors.[4] In addition to the various types of magnetic materials, there is also a variety of magnetic functional materials with various niche functions and applications such as giant magnetic resistors, magneto-strictive materials, magnetic fluids and magnetic refrigeration materials. For these applications pure

organic, pure inorganic and hybrid organic–inorganic composite materials have become of increasing interest.

Within the past few decades many classic bulk materials (such as magnetic materials) have been processed into shapes with one or more dimensions at the nanoscale, rendering the materials with desirable properties that nanotechnology delivers. Among these magnetic nanomaterials, materials with two-dimensional scale constraints such as nanofibres (NFs) achieve incredible advances due to their anisotropic nature. NFs exhibit great enhancement and control of many properties with the most notable being flexibility, large specific surface area, porosity and coercivity (H_c).

Electrospinning is a simple means of processing materials to form NFs, where polymer chains align themselves under an electrostatic force to form elongated, thin, filamentous nano-structures. During the process, a polymer solution (or melt) is stretched and deformed by the electrostatic force and a droplet forms at the tip of needle. The shape of the droplet is determined by gravity, viscosity, surface tension and electric field. In the process of electrospinning, the most common droplet shape is a cone, referred to as the Taylor cone.[5,6] When charge repulsion exceeds surface tension the polymer is pulled from the end of the Taylor cone. The modes of flow are also determined by the

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aforementioned forces. The withdrawn flow initially experiences stable motion and is then forced into an unstable stage where the polymer solidifies in the air to form fibres, which are then received by the collector.

The final properties of the fibres are influenced by three key factors:

(i) solution properties (such as the viscosity, concentration, polymer molecular weight and dielectric properties of the solution);

(ii) processing parameters (such as applied voltage, needle-to-collector distance and feeding rate) and

(iii) environmental conditions (such as temperature, humidity and air flow around the system).

These factors demonstrate the diverse range of magnetic electrospun NFs that can be produced from a single chemical composition; as the structure of a single NF and the fibre assembly can be manipulated by other means. Indeed, advances in the control over electrospun NFs is envisioned to vastly benefit the magnetic material field. For example, electrospinning is the only known method used to prepare continuous ultra-long, thin NFs.[7]

To the best of our knowledge, there are no comprehensive reviews of electrospun magnetic NFs that evaluate their methods of production, properties and applications. Herein, we have reviewed articles from 20 years of research on electrospun magnetic fibres. The review is divided into three main parts (as shown in Fig. 1): (i) electrospinning organic–inorganic hybrid magnetic materials (summarised in Table 1); (ii) using electrospun fibres as templates for the creation of both hybrid and solely inorganic magnetic materials (summarised in Table 2); and (iii) applications of magnetic nanofibrous materials. As explained in this review, organic–inorganic composite magnetic nano-materials can be prepared via a one-step method and the resulting nanofibrous matrix can provide magnetic nanoparticles (MNPs) with mechanical support, protection against oxidation and favourable dispersion (Section 2). Alternatively, for pure inorganic

magnetic NFs (Section 3), electrospinning is a simple, available tool used in the fabrication of fibrous templates with different morphological structures enabling the user to manipulate the magnetic properties of the final product. The major difference between these two strategies is that organic matter is removed in the latter to create the final inorganic product. In the final section (Section 4), we explore the various applications of these advanced materials from electromagnetic interference (EMI) shielding and pollutant treatment, to biomedical devices in drug delivery and tissue engineering. In short, this review focuses on the processing methods of electrospun magnetic composite NFs and pure inorganic NFs from templates, their properties and related applications.

Finally, as aforementioned, we summarise the complete literature of magnetic NFs in two comprehensive tables, where Table 1 covers systems in the category described in Section 2 [i.e. organic–inorganic hybrid magnetic nanofibrous materials created directly from electrospinning solution(s)] and Table 2 lists those systems created from electrospun fibrous templates, as described in Section 3. Both tables describe the methodological approaches used, properties obtained and potential applications, where relevant. We anticipate that these tables will serve as a useful repository for researchers, in addition to those new to the field, looking to study such advanced materials.

2. Electrospinning magnetic materials: reagents, methods and morphologies

Electrospun polymer–magnetic material composite fibres have been widely reported across the literature. This section discusses the various electrospinning methods and parameters used to manipulate the creation of magnetic NFs, before reviewing the variety of magnetic materials that have been used. Table 1 serves as a comprehensive summary of this section, listing the various systems that have been explored, the magnetic reagent(s), polymer(s), electrospinning approach, major magnetic property and applications (where relevant) in each case.

2.1 Electrospinning methods and fibre structures

Electrospinning is a primary method used to prepare continuous nanofibrous materials with advantages such as simple device manufacture, material compatibility and controllable fibre morphology. Fibres produced by electrospinning have high surface area-to-volume ratios, tuneable surface morphology and controlled alignment. The specific morphology of the fibre can control the distribution of nanoparticles (NPs).[8] In this section, we discuss the various electrospinning parameters that can be used to create an array of NFs, specifically focusing on magnetic NFs.

2.1.1 Materials. NFs can be produced by electrospinning from a solution or melt of a variety of materials. Adequate entanglement is the key for the material to be suitable for electrospinning. Most polymeric materials produce enough entanglement to perform both solution and melt electrospinning. The extent of entanglement is related to the polymer molecular weight, the concentration and the solvent system. Sufficiently low molecular weight, low
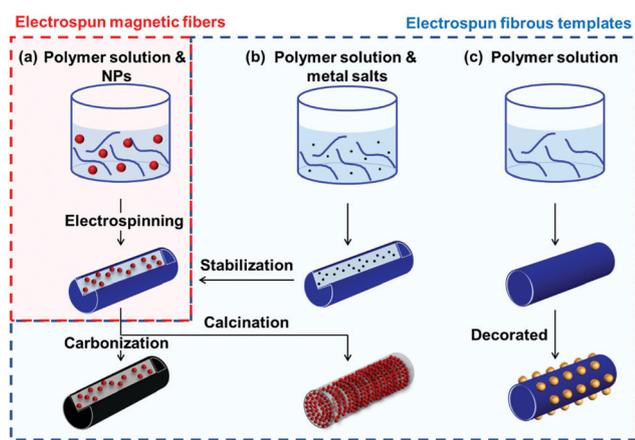
Fig. 1 General methods to prepare electrospun magnetic fibres, where the area process highlighted in red is discussed in Section 2 (a) and the processes highlighted in blue are discussed in Section 3 (b and c). (a) Direct method for producing magnetic nanofibres; (b) templating procedure of MNFs from a magnetic pre-cursor solution and (c) templating procedure of MNFs where the magnetic component is deposited upon them.

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Table 1 Organic–inorganic hybrid magnetic nanofibrous materials created directly from electrospinning solution(s) discussed in Section 2, detailing magnetic components and polymers used, electrospinning method followed, the magnetic properties of the resulting materials and their applications, where relevant

Magnetic ingredients	Polymer	Electrospinning approaches	Magnetic properties	Applications	Ref.
Fe$_2$O$_3$	PVP	Uniaxial – blend	H_c = 327 Oe, M_r/M_s = 0.244		44
	PMMA, PU	Uniaxial – blend	Superparamagnetic, M_s = 6.172 emu g^{-1}		37 and 90
	PVA	Uniaxial – deposition		EWA	289
γ-Fe$_2$O$_3$	PLA	Uniaxial – blend	Paramagnetic, superparamagnetic, M_s = 0.049 emu g^{-1}	Cell culture	88, 89, 254 and 255
Fe$_3$O$_4$	PVA	Coaxial	Paramagnetic, superparamagnetic	Tissue engineering scaffolding; Oil adsorption/separation, cell culture	290
	CA	Uniaxial – blend	Superparamagnetic, M_s = 2.284 emu g^{-1}	Water treatment	253
	Cellulose–CS, PEO	Uniaxial – blend	Superparamagnetic, M_s = 18.61 emu g^{-1} (max)	Fluorescence self-display and adsorption removal of mercury(II)	59
	Cellulose	Coaxial spinning – post treatment			60
	Cellulose pulp	Uniaxial – blend	Ferromagnetic	Nanofibrous scaffolds	291 and 292
	CS	Uniaxial – blend	Superparamagnetic and ferromagnetic at different temperature	Hyperthermia treatment of tumor cells	145
	CS, PEO	Uniaxial – blend	Superparamagnetic, M_s = 8.47–7.12 emu g^{-1}	Adsorption removal of heavy metal	293
	CMC, PVA	Uniaxial – blend	Superparamagnetic, M_s = 11.21–16.94 emu g^{-1} (RT)	Hypothermic tumor cell treatment	294
	CS, PVA	Uniaxial – blend	Ferromagnetic, H_c = 216–222 Oe	Bone regeneration	295
	CS : PVA = 3 : 7	Uniaxial – blend	Superparamagnetic, M_s = 0.67–3.19 emu g^{-1}	Chromium(VI) removal	296, 22
	CTMB	Uniaxial – blend	Superparamagnetic, M_s = 20.98 emu g^{-1}	Enantioselective adsorption of racemic drug	297
	DNA–CTMA	Uniaxial – blend	Superparamagnetic	Water detoxification	298
	Gelatin	Uniaxial – blend	Superparamagnetic, M_s = 3.05–12.87 emu g^{-1}		20
	HPMCP, CA	Uniaxial – blend	Superparamagnetic, M_s = 0.28–0.52 emu g^{-1}	Drug release	18
	MADO	Uniaxial – blend	Superparamagnetic, M_s = 83.9–74.6 emu g^{-1}	Hypothermic chemotherapy	299
	P(AN-co-AA)	Uniaxial – blend	Superparamagnetic, M_s = 27.02–30.51 emu g^{-1}		300
	PA6	Uniaxial – blend	Superparamagnetic, M_s = 1.03 emu g^{-1}	EMS	301 and 302
	PAAm, PVA	Uniaxial – blend	Superparamagnetic, M_s = 16.6, 47.7, 48 emu g^{-1}, M_r = 2.8 ± 0.5 emu g^{-1}, H_c = 150 Oe		303
	PAN	Uniaxial – blend	Superparamagnetic, H_c = 20.1–206.7 Oe, M_s = 4.67 emu g^{-1}	Magnetic separation of the photo-sensitizers, electrets filter media	21 and 304–306
			M_s = 3.5–8.4 emu g^{-1}	Cell separation, drug targeting	307
			Ferromagnetic	Microwave absorption	308
			Superparamagnetic, M_s = 81.389 emu g^{-1}	Phenol removal	38
			M_s = 8.6 emu g^{-1} (max)	Separation of glycoproteins	309
		Twisted blend	Superparamagnetic, M_s = 10.10–28.77 emu g^{-1}		33
		Layer-by-layer	Superparamagnetic, M_s = 5.56–20.76 emu g^{-1}		36
	PAN-co-AA	Uniaxial – blend	Superparamagnetic, M_s = 3.6 emu g^{-1}	Oil adsorption/separation	310
		Uniaxial – blend		Adsorbents for removal of malachite green from water and wastewaters	311
	PANI, PAN	Layer-by-layer	M_s = 10.848–21.856 emu g^{-1}		67
	PANI, PVP	Uniaxial – blend	Superparamagnetic, M_s = 3.35–10.89 emu g^{-1}	EMS	16
	PANI/PMMA	Coaxial	M_s = 4.52 emu g^{-1}		66
	PBT	Uniaxial – blend	Paramagnetic	Thin film microextraction, magnetic separation	312 and 313
	PCL	Uniaxial – blend	Weak ferromagnetic or superparamagnetic, H_c ~ 2.5 Oe, M_r ~ 0.27 emu g^{-1}, M_s = 1.0–11.2 emu g^{-1}	Magnetically-actuated, electro-magnetic heating; Tissue engineering scaffold	314, 256

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Table 1 (continued)

Magnetic ingredients	Polymer	Electrospinning approaches	Magnetic properties	Applications	Ref.
			$M_s = 27.7-103.9$ emu g^{-1}, $H_c \sim 70$ Oe	Organic pollutants degradation	315
			Superparamagnetic	Drug delivery vehicle	316
			Ferrimagnetic, $M_s = 88.5$ emu g^{-1}, $H_c = 80$ Oe	Magnetic heating	58
		Coaxial	Superparamagnetic		317
		Uniaxial – blend–UV cross-linking	Superparamagnetic, $M_s = 71.549$ emu g^{-1}		62
	PCL:CS = 6:1	Coaxial	Superparamagnetic, $M_s = 0.74-3.52$ emu g^{-1},	Drug release	318
		Uniaxial – blend	$H_c = 13.25-17.10$ Oe	Hyperthermia	319
	PEK-C	Uniaxial – blend–thermal treatment		EWA	320
	PEO	Coaxial			42
	PEO, PVA	Janus	Superparamagnetic		321
	PEO/PLLA	Uniaxial – blend	Superparamagnetic	Malachite green adsorption	322
	PEO/PVP	Uniaxial – blend			323
	PET	Uniaxial – blend	$M_s = 0.58-2.79$ emu g^{-1}, $M_r = 0.1-0.41$ emu g^{-1}, $H_c = 79.94-103.9$ Oe	EMS	12
			Ferromagnetic, near-superparamagnetic		71
		Coaxial	Superparamagnetic		43
	PF–Na, PVA	Uniaxial – blend	Superparamagnetic, $M_s = 9.7$ emu g^{-1}		324
	PHB	Uniaxial – blend	$M_s = 2.4-4.9$ emu g^{-1}	Photocatalyst	325
	PHB, PHVB	Uniaxial – blend	Superparamagnetic, $M_s = 0.42-2.51$ emu g^{-1}		326
	PHEMA, PLLA	Uniaxial – blend	Superparamagnetic		327
	PLA	Uniaxial – blend	Paramagnetic	Electromagnetic heating	31
	PLA, PCL	Uniaxial – blend	$M_s = 1.26-3.37$ emu g^{-1}	Drug delivery	268
	PLA, PEG	Uniaxial – blend	Superparamagnetic, $M_s = 3.57-10.07$ emu g^{-1}	Smart clothing	328
	PLGA	Uniaxial – blend	$M_s = 1.37-3.94$ emu g^{-1}, paramagnetic, superparamagnetic	Tissue engineering scaffolds	85
	PLLA	Uniaxial – blend	Superparamagnetic, $M_s = 5.22-23.19$ emu g^{-1}	Tissue engineering scaffold	257 and 329
	PMMA	Uniaxial – blend	$M_r = 8.38$ emu g^{-1}, $M_s = 4.23-35.77$ emu g^{-1}		55 and 330
		Coaxial	$M_s = 2.96-32.61$ emu g^{-1}, superparamagnetic		63, 64, 74 and 331
		Janus	$M_s = 31.98$ emu g^{-1}		30, 65, 86 and 332–335
		Janus – coaxial	$M_s = 3.6-24.0$ emu g^{-1}		336
		Uniaxial – blend – cospinning			337
	PMMA, PANI	Uniaxial – blend – cospinning	$M_s = 7.69$ emu g^{-1}		338
	PMMA/PANI	Janus	Superparamagnetic, $M_s = 23.52$ emu g^{-1}		28
	PNIPAM	Uniaxial – blend – cross-linking	Superparamagnetic, $M_s = 7.88$ and 15.80 emu g^{-1}		339
	Polythiophene, CS	Uniaxial – blend		Solid-phase extraction of triazine herbicides	340
	PS	Uniaxial – blend	Superparamagnetic, $M_s = 1.18-3.66$ emu g^{-1}, coercivity = 6.14–8.98 Oe, retentivity = 1.8–8.4 Oe, $M_s = 2.77$ emu g^{-1}, $M_s = 2.42$ emu g^{-1}	Remote and efficient oil adsorption	57 and 341
	PS, PVDF	Two-nozzle blend	Ferrimagnetic, $M_s = 26$ emu g^{-1}, $M_r = 10$ emu g^{-1}, $H_c = 20$ Oe	Cancer therapy	34
	PVA	Uniaxial – blend		Water oil separation	46, 51 and 342
					61
		Twisted blend	Superparamagnetic, $M_s = 7.11-21.28$ emu g^{-1}		56

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Table 1 (continued)

Magnetic ingredients	Polymer	Electrospinning approaches	Magnetic properties	Applications	Ref.
	PVA, guar gum	Uniaxial – deposition	Superparamagnetic, $M_s = 0.1$–5.8 emu g^{-1}	Wastewater treatment	94
	PVA, PAA	Uniaxial – blend	Ferromagnetic, $M_s = 1.72$–6.77 emu g^{-1}		72
	PVA : PAA = 5 : 6	Uniaxial – blend	Superparamagnetic, $M_s = 10.9$–39.9 emu g^{-1}	Tissue engineering scaffolds	343
	PVA, PCL	Uniaxial – blend		Tissue engineering scaffolds	258
	PVA–PHB/PCL	Coaxial	$M_s = 1.9 \pm 0.3$ emu g^{-1}	Photocatalyst	344
	PVC	Uniaxial – blend	Superparamagnetic, $M_s = 1.93$–12.5 emu g^{-1},	Microwave absorption	345
	PVDF	Uniaxial – blend	$H_c = 96$–113 Oe, $M_s = 46.5$ emu g^{-1}	Triboelectric nanogenerator	19, 282 and 346
			Superparamagnetic	Hyperthermia treatment and skin wound healing applications	347
	PVP	Coaxial	Superparamagnetic, $M_s = 1.76$–13.59 emu g^{-1}		26, 68, 70 and 348–350
		Janus	$M_s = 1.73$–18.99 emu g^{-1}		27, 35, 47, 87, 351 and 352
			Superparamagnetic, $M_s = 2.63$–10.19 emu g^{-1}		14, 69, 83, 84 and 353
Fe$_3$O$_4$, γ-Fe$_2$O$_3$		Janus – uniaxial – blend	$M_s = 7.4$–14.84 emu g^{-1}	EWA	354
		Uniaxial – blend	$M_s = 70.2$ emu g^{-1}, $M_r = 8.7$ emu g^{-1}, $H_c = 91$ Oe		355
	PVP/PLLA	Uniaxial – blend	Superparamagnetic, $M_s = 36.6$ emu g^{-1}		356 and 357
	Silk fibroin	Uniaxial – blend & coating	Superparamagnetic		73
	PS-b-PI	Coaxial	Superparamagnetic, $M_s = 60$ emu g^{-1}	Tissue engineering scaffolds	8
			Superparamagnetic (>13 kOe), $H_c = 250$ Oe (5 kOe)	Tissue engineering scaffolds	91
	P(NIPAM-co-HMAAm)	Uniaxial – blend & thermal crosslinking		Induction of skin cancer apoptosis	259
Fe$_3$O$_4$, Fe$_2$O$_3$/NiO	PAN	Uniaxial – blend	$M_s = 30.6$ emu g^{-1} (max), $H_c = 188.2$ Oe (max)	Data storage and transfer	1
Fe–FeO	PI	Uniaxial – blend	Superparamagnetic, $M_s = 18.84$ emu g^{-1}		50
IONPs	PLGA	Uniaxial – blend	Soft ferromagnetic, $M_s = 0.53$ emu g^{-1}	Hyperthermia treatment and controlled drug release	358
	PVP	Uniaxial – blend	Superparamagnetic, $M_s = 6.8$ emu g^{-1}		266
	PCL	Uniaxial – blend	Superparamagnetic	Mesenchymal stem cell proliferation	359
SPIONs	PDLLA	Uniaxial – blend	Superparamagnetic		360
CoFe$_2$O$_4$	PAN	Uniaxial – blend & stabilization	Superparamagnetic, $M_s = 50$ emu g^{-1}, $T_B = 125$ K		361
Nd$_{0.05}$Bi$_{0.95}$Fe$_{0.95}$Co$_{0.05}$O$_3$	PVDF–TrFE	Uniaxial – blend	Ferromagnetic, $M_r = 0.58$ emu g^{-1}, $H_c = 1400$ Oe		362
Magnetic bioglass	PVA	Uniaxial – blend	Weak soft ferromagnetic, $H_c \sim 20$ Oe, $M_r \sim 0.01$ emu g^{-1}	Bone scaffolds	260
FePt	PCL	Coaxial	Superparamagnetic		101 and 363
Mg-ferrite	PCL	Uniaxial – blend	Ferromagnetic, $M_s = 0.024$–3.19 emu g^{-1}	Enhanced cell attachment, growth and proliferation	364
Magnetic zeolite	PAN	Uniaxial – blend	$M_s = 15$ emu g^{-1}, coercivity = 97 Oe	Determination of polycyclic aromatic hydrocarbons in water samples	365
Ni	PS	Uniaxial – blend	$M_s = 0.08$–1.52 emu g^{-1}	Remove arsenic from water	98
SrFe$_{12}$O$_{19}$	PVA	Uniaxial – blend	$M_r/M_s = 0.72$, $H_c = 6.3$ kOe		17
Gd(DTPA)	Eudragit S100, PEO	Coaxial		Drug delivery and MRI imaging	265
SrTiO$_3$/NiFe$_2$O$_4$	PVP	Uniaxial – blend & side-by-side uniaxial	Ferromagnetic, $M_s = 10$–18 emu g^{-1}		29
Fe-Doped In$_2$O$_3$/α-Fe$_2$O$_3$ (porous nanotubes & particle-in-tubes)	PVP	Coaxial	$M_s = 0.48$–22.37 emu g^{-1}		366
CeO$_2$–γ and CoFe$_2$O$_4$	PVP	Uniaxial – blend	$H_c = 320.40$–541.35 Oe, $M_s = 18.07$–17.03 emu g^{-1}, $M_r = 3.72$–4.04 emu g^{-1}	Solar light driven photocatalyst	367
Mixture of magnetite (Fe$_3$O$_4$) and maghemite (γ-Fe$_2$O$_3$)	P(NIPAM-co-HMAAm)	Uniaxial – blend		Inducing cancer apoptosis	1

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Table 1 (continued)

Magnetic ingredients	Polymer	Electrospinning approaches	Magnetic properties	Applications	Ref.
$SrRE_{0.6}Fe_{11.4}O_{19}$ (RE = La, Ce)	PVP	Sol–gel & uniaxial – blend	H_c = 4890.3–5321.9 Oe, M_s = 53.475–53.839 emu g^{-1}, M_r = 28.517–28.765 emu g^{-1}		368
$BaFe_{12}O_{19}$	PVP	Uniaxial – blend & coaxial	M_s = 45.03–55.59 emu g^{-1}, M_r = 1.2–27.12 emu g^{-1}, H_c = 102.01–3761.57 Oe		369
$CoFe_2O_4@Y_2O_3$:5%Tb$_3$	PVP	Uniaxial & coaxial	M_s = 20.05–20.67 emu g^{-1}		370
$NaYF_4$:Eu^{3+} and Fe_3O_4	PVP	Uniaxial – blend	Superparamagnetic, M_s = 3.87–16.99 emu g^{-1}		371
Fe	PVP	Uniaxial – blend			372
	PANI	Uniaxial – blend		Magnetic hypothermia treatment	373
Co	CA	Uniaxial – blend		Stem cells osteogenic differentiation	102
MnZnFe–Ni nanoparticles	Estane	Uniaxial – blend	Superparamagnetic, M_s = 1.67–25 emu g^{-1}		99
MGNPs	Beta-lactoglobulin, PEO	Uniaxial – blend	M_s = 0.1–4.16 emu g^{-1}, M_r = 0.01–1.25 emu g^{-1}, H_c = 89–114 Oe		78
Ni	Polycarbonate-urethanes	Uniaxial – blend	Ferromagnetic		374
$CoFe_2O_4$, Fe_3O_4	Silk	Uniaxial – blend			375
YFe garnet, YGdFe garnet	CA	Uniaxial – blend	Hard magnetic		376
Strontium hexaferrite	PVA	Uniaxial – blend		Bio-separation	76
TiO_2/SiO_2	PAN, PVP	Uniaxial – blend		EMS	377
MWCNTs	PVA	Uniaxial – blend		EMS	378
	PVP	Uniaxial – blend		EMS	379 and 380
$FeCl_3$	PVA	Uniaxial – blend			381
Ferritin	PVA	Uniaxial – blend	Superparamagnetic	Artificial muscles, MRI	100

concentration or poor solubility typically result in beaded structures (beaded fibres or discrete microspheres) rather than long defect-free fibres. Wang et al.[9] studied the effect of solvent on the preparation of fibres and beads and found that decreasing the compatibility between the solvent and the polymer decreased the intermolecular entanglement, thus producing beads.

Nanoparticles can also be electrospun by adding nanoscale components into the polymer carrier. However, in order to form uniform nanoparticles some materials require pre-treatment. Common examples of pre-treatment include sol–gel treatment of the polymer solution and dispersive pre-treatment of composite nanoparticles, such as ultrasonic dispersion[10–13] or coating with oleic acid (or grafting with a coupling agent) to prevent undesirable aggregation.[14–17]

2.1.2 Electrospinning rigs. Fig. 2 shows the wide range of electrospinning rigs that have been used to create various nanofibrous fabrics. The most commonly used electrospinning rig is relatively simple and is comprised of a power supply, a liquid supply device and a collector (Fig. 2(a)). The power supply provides sufficient electric field force to create liquid filaments. Adjustment of the liquid supply parameters such as; flow rate of solution, needle structure and mode of liquid supply, allow for continuous and uniform fibres to be formed. In addition to common single and multiple needle set-ups, there are also needleless devices such as silk thread, bulged roller and liquid pool. In these cases, the collector is usually a flat plate and the morphology of the fibre mat can be regulated by changing the collection method.[11]

2.1.2.1 Uniaxial electrospinning. Uniaxial electrospinning is the most simplistic device design (Fig. 2(a)). The liquid supply uses a solitary nozzle to prepare a single component of solid micro/nanofibre.[18] Fibres have been prepared, via uniaxial electrospinning, with different morphologies including; nano-belt (Fig. 3(c)), bead-on-string (Fig. 3(d)), and connected fibre mesh structures.[19–22] The various morphologies provide different characteristics, such as hydrophobicity and mechanical properties. Superhydrophobicity is caused by the huge specific surface area of the fibrous mesh membrane. The large surface area of the membrane can greatly reduce the contact area between the fibrous membrane surface and the liquid. Additionally, the micro- and nanoscale voids on the surface of the specimen can easily trap air, and when water droplets come into contact with the material the air holds the droplets up in accordance with the typical Cassie–Baxster contact model.[23] Additionally, it has been shown that electrospinning can be used to construct a composite structure of electrospun nanofibres decorated with microspheres to provide the superhydrophobic character. On this basis it becomes easier to design and control the micro–nano structure of the surface, allowing the perfect realisation of superhydrophobic structures.[24,25]

2.1.2.2 Coaxial electrospinning. Coaxial electrospinning (Fig. 2(b and c)) can be used to produce continuous, single-channel, or multi-channel core–shell and hollow fibres. The fibres are prepared by using two or more coaxial nozzles of

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Table 2 List of magnetic materials (both hybrid and solely inorganic) fabricated using a template approach discussed in Section 3, listing magnetic reagents, polymer(s) used, electrospinning and post-treatment procedures followed, morphology and properties of the final material obtained and applications where relevant

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
Fe	PVP	Uniaxial – carbonisation	Regular	M_s = 85 emu g^{-1}, H_c = 526 Oe	EWA	232
		Multi-nozzle – calcination – reduction	Regular		EWA	382
	PAN	Uniaxial – carbonisation	Porous 3D cross-linked network	Ferromagnetic, M_s = 32.95 emu g^{-1}, H_c = 406 Oe	EWA	226
		Uniaxial – carbonisation–activation	3D non-woven network	Ferromagnetic, M_s = 11.2–22.2 emu g^{-1}	Catalyst for PMS activation	383
Co	PVA	Uniaxial – heating	Regular	H_c = 0–275.0 Oe		120
	PVP	Uniaxial – stabilisation – carbonisation – mixed with paraffin and pressed into toroidal shaped specimens	Regular	Soft-magnetic, M_s = 73 emu g^{-1}	EWA	227
	PVP & PAN	Uniaxial – calcination – reduction	Necklace-like	Ferromagnetic, M_s = 28.37 emu g^{-1} (max), H_c = 674–1016 Oe		384
		Uniaxial – stabilisation – carbonisation	Regular	M_s = 60 emu g^{-1}	Catalysts	385
	PAN	Uniaxial – calcination	Fibres with deposited particles	Ferromagnetic, M_s = 10.1–23.9 emu g^{-1}, H_c = 504.6–701.1 Oe	Microwave absorption	386
	TEOS/PVA	Uniaxial – carbonisation	Regular	M_s = 15 emu g^{-1}	Catalyst for AR	387
		Uniaxial – calcination – lyophilisation – reduction – coating	Fibres with deposited particles	M_s = 27.1 emu g^{-1} (max)	Absorbents	388
	PVA	Uniaxial – carbonisation	Regular	Ferromagnetic, M_s = 77.52 emu g^{-1}, H_c = 261.3 Oe, M_r = 7.97 emu g^{-1}, H_s = 8500 emu g^{-1} (RT), M_s = 78.45 emu g^{-1}, H_c = 392.7 Oe, M_r = 9.26 emu g^{-1}, H_s = 8500 emu g^{-1} (5 K)		389
Ni	PVA	Uniaxial – calcination	Regular	Ferromagnetic, M_s = 25.3 emu g^{-1}, H_c = 382.52 Oe, M_r = 10.4 emu g^{-1}, H_s = 3700 emu g^{-1} (5 K), M_s = 23.12 emu g^{-1}, H_c = 67.66 Oe, M_r = 3.06 emu g^{-1}, H_s = 800 emu g^{-1} (RT)		390
		Uniaxial – calcination – deoxidation	Regular	M_s = 51.9 emu g^{-1}, H_c = 185 Oe, M_r = 16.9 emu g^{-1}		127
	PAN	Uniaxial – calcination	Porous		Electromagnetic interference shielding	391
	PAN & PDA	Uniaxial – carbonisation	Regular		Lithium-ion batteries	392
		Uniaxial – carbonisation – surface modification	Short		Biosensors	393
Fe$_3$O$_4$	PAN	Uniaxial – stabilisation – carbonisation – mix with paraffin	Regular	M_s = 3.32–13.53 emu g^{-1}, H_c = 1.21–274 Oe, M_r = 0.05–1.8 emu g^{-1}	EWA	196
		Uniaxial – stabilisation – carbonisation	Regular	Ferromagnetic, M_s = 30 emu g^{-1}, M_r = 2.69 emu g^{-1}, H_c = 189 Oe	EWA	197
		Uniaxial – stabilisation – carbonisation – coating	Core-shell	M_s = 4–7.5 emu g^{-1}, H_c = 180–240 Oe, M_r = 1.49–2.62 emu g^{-1}	EWA	150
		Uniaxial – hydrothermal – deposition	Nanofibres with deposited particles		Adsorbents for water purification	112
		Coaxial – carbonisation	Regular	M_s = 10.18–39.65 emu g^{-1}	Magnetic, electronic and bio-applications	161
		Coaxial – carbonisation	Core-shell	M_s = 6.1–81.2 emu g^{-1}		15
	PAN/BA-a	Uniaxial – stabilisation – calcination	Hierarchical porous	H_c = 71 Oe, M_r = 0.654 emu g^{-1}	Absorbents for organic dyes in water	394
	PAN/PMMA	Uniaxial – calcination	Porous nanobelts	M_s = 5.54–18.49 emu g^{-1}	Absorbents for organic dyes	111
	PAN–PEI	Uniaxial – calcination	Sintered particles to form a fibre	Superparamagnetic, M_s = 78.79 emu g^{-1}, M_r = 0.528–10.5 emu g^{-1}	Absorbent	395

Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
	PVP	Uniaxial – stabilisation – calcination – mix with bismaleimide	Regular		EWA	148
		Uniaxial – calcination – reduction	Necklace-like	Ferromagnetic, M_s = 58.4 emu g^{-1}, H_c = 186.7 Oe	EWA	396
		Uniaxial – calcination – reduction	Smooth	M_s = 57.6 emu g^{-1}, H_c = 188.4 Oe, M_r/M_s = 0.28		131
		Uniaxial – oxygen plasma treatment	Sintered particles to form a fibre	Ferromagnetic, M_s = 72.4–35.51 emu g^{-1},		41
	PEO	Uniaxial – calcination	Short fibres with deposited nanoparticles	M_r = 3.15–0.07 emu g^{-1}, H_c = 24.59–1.35 Oe	EWA	397
	PBZ	Uniaxial – calcination	Sintered particles to form a fibre	M_s = 5.95–9.22 emu g^{-1}, superparamagnetic	Water treatment	398
	PVDF	Uniaxial – deposition	Nanofibres with clusters		Remote controllable oil removal	399
	PU	Uniaxial – deposition	Fibres with deposited particles	Superparamagnetic, M_s = 33.12 emu g^{-1}	Hyperthermia treatment	107
	PEO & PVA	Uniaxial – crosslinking – co-precipitation	3D cross-linked network with deposited particles	Superparamagnetic, M_s = 9–18 emu g^{-1}, T_B = 70–75 K	Hyperthermia treatment	109
	CS	Uniaxial – deposition	3D cross-linked network with deposited particles	Superparamagnetic, M_s = 16.3–27.2 emu g^{-1} (300 K), ferromagnetic, H_c = 284–298 Oe, M_r = 4.9–7.9 emu g^{-1} (10 K)	Hyperthermia treatment of tumour cells	145
	PVA	Uniaxial – carbonisation	Regular	Ferromagnetic, M_s = 50.27–62.8 emu g^{-1}, H_c = 50.2–150.72 Oe		400
	PVA	Uniaxial – anneal – deposition	3D cross-link network			168
	PVA/PAA	Uniaxial – blend – thermal treatment – deposition	Cross-linked mat with deposited particles	Superparamagnetic, M_s = 32.5 emu g^{-1}	Recyclable catalytic capacities	401
	PAA	Uniaxial – thermal treatment	Fibres with deposited particles	M_s = 1.52–10.46 emu g^{-1}		187
	PANI	Uniaxial – deposition	Nanofibres with deposited particles	Ferromagnetic, M_s = 1.9 emu g^{-1}, H_c = 930 Oe, M_r = 25.83 × 10^{-3} emu g^{-1}		108
α-Fe$_2$O$_3$	PVA	Uniaxial – deposition – calcination	Hollow	Weak ferromagnetic, M_s = 24.4 emu g^{-1}	Absorbents for dyes	162
		Uniaxial – calcination	Regular	M_s = 20 emu g^{-1}, H_c = 40 Oe (α-Fe$_2$O$_3$)	Catalyst for azo dyes degradation	245
		Uniaxial – calcination	Nanorod	Superparamagnetic–ferromagnetic		188
	PVP	Uniaxial – calcination	Nanotubes	Ferromagnetic, M_s = 0–20 emu g^{-1}, H_c = 150–760 Oe, permanent magnetic material	Supercapacitor electrodes	217
		Uniaxial – calcination	Nanotubes	Ferromagnetic, H_c = 256.71–628.18 Oe, M_r = 0.2872–0.4512 emu g^{-1}		163
γ-Fe$_2$O$_3$	PLGA & PCL	Uniaxial – deposition	Nanofibres with deposited particles	Superparamagnetic, M_s = 3.56 emu g^{-1}	Stem cell differentiation	144
	PVA	Uniaxial – hydrothermal synthesis deposition – calcination	Core-shell	Ferromagnetic, M_s = 98.9 emu g^{-1}, H_c = 175.5 Oe, M_r = 6.9 emu g^{-1}	Sensors	137
		Uniaxial – calcination	Regular	Ferromagnetic, M_s = 52.1 emu g^{-1}, H_c = 460 Oe, M_r = 16.7 emu g^{-1}, H_s = 6000 Oe (5 K); M_s = 45.2 emu g^{-1}, H_c = 218 Oe, M_r = 11 emu g^{-1}, H_s = 2000 Oe (RT)	Semiconductor	402
	PVP	Uniaxial – calcination	Core-shell fibres, fibre-in-tube, tube-in-tube	Ferromagnetic, M_s = 55.2 emu g^{-1} (fibre-in-tube), M_s = 56.3 emu g^{-1} (tube-in-tube), M_r = 5.5–15.9 emu g^{-1} (tube-in-tube), H_c = 78–206 Oe, M_r/M_s = 0.1–0.28		403
Fe$_2$O$_3$	PVP	Uniaxial – calcination	Regular	Superparamagnetic–antiferromagnetic M_s = 0.6 emu g^{-1}	Catalyst	177
		Uniaxial – calcination – post treatment	Porous hollow fibres with nanoflakes coating		Photocatalyst	237
		Uniaxial electrospinning – calcination	Sintered particles to form a fibre	Superparamagnetic, M_s = 8.42 emu g^{-1}, M_r = 1.2 emu g^{-1}, H_c = 160 Oe	Photocatalyst	236

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Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
Fe_xO_y	PVP	Uniaxial – calcination	Porous nanosheets, nanotubes	Ferromagnetic, M_s = 2.84–18.91 emu g^{-1}, H_c = 106.27–152.87 Oe		105
$\alpha\text{-}Fe_2O_3$, Fe_3O_4	PAA	Uniaxial electrospinning – deposition	Nanofibres with deposited particles	Superparamagnetic, M_s = 2.8–4.0 emu g^{-1}		143
$\alpha\text{-}Fe_2O_3$, Co_3O_4	PVP	Uniaxial – calcination	Hollow	M_s = 0.75–1.50 emu g^{-1}	EWA	103
Co_3O_4	PAN	Uniaxial – stabilisation – carbonisation	Regular	Soft ferromagnetic		121
NiO	PVP	Uniaxial – calcination	Sintered particles to form a fibre	M_s = 55.3 emu g^{-1}, H_c = 194.0 Oe		404
NiO	PEtOx	Uniaxial – calcination	Sintered particles to form a fibre		Alcohol sensor	405
CuO/NiO	PVA	Uniaxial – calcination	Regular	Ferromagnetic, H_c = 107.3 Oe (max), M_r = 0.472 emu g^{-1} (max)		203
ZnO	PVA	Uniaxial – calcination	Regular	Paramagnetic, M = 0.337–0.480 emu g^{-1}, H_c = 10 kOe		406
ZnO	PVA	Uniaxial – anneal	Regular	Ferromagnetic, M_s = 0.039 emu g^{-1} (max)		192
ZrO_2	PVP	Coaxial electrospinning – anneal – deposition	Hollow		Catalysts	166
	PVP	Uniaxial – anneal	Nanofibres with smooth surface	M_s = 0.45–0.57 emu g^{-1}	Photocatalysts for the degradation of organic pollutes	241
SnO_2	PVP	Uniaxial – anneal	Nanotubes	M_s = 0.012–0.017 emu g^{-1}, H_c = 79–95 Oe (300 K), M_s = 0.11–0.19 emu g^{-1}, H_c = 163–169 Oe (5 K)		173
Fe_3C	PAN	Uniaxial – carbonisation	Short	Soft ferromagnetic, M_s = 18.0 emu g^{-1}, H_c = 108.3 Oe, M_r = 0.68 emu g^{-1}	EWA	228
Fe_3O_4, $\alpha\text{-}Fe_2O_3$, Fe_2N	PVP	Uniaxial – calcination	Sintered particles to form a hollow fibre	Fe_3O_4: M_s = 82.99 emu g^{-1}, M_r = 39.27 emu g^{-1}, H_c = 400.45 Oe $\alpha\text{-}Fe_2O_3$: M_s = 4.34 emu g^{-1}, Fe_2N: M_s 2.07 emu g^{-1}	High performance anodes for LIBs	134
FeC_3/FeN_3	PVP	Uniaxial – stabilisation – nitridation	Branch-like (400–700 °C)	Ferromagnetic, M_s = 122 emu g^{-1} Fe, H_c = 112 Oe Fe, M_r = 10.4 emu g^{-1} Fe		407
Fe@FeO	PAN	Uniaxial – stabilisation – carbonisation	Short	Ferromagnetic, M_s = 6.8–24.1 emu g^{-1}, H_c = 45–209 Oe		408
Pd doped Co	PVA	Uniaxial – calcination	Regular		Photocatalyst	409
Fe-Doped NiO	PVA	Uniaxial electrospinning – calcination	Regular	Ferromagnetic	Diluted magnetic semiconductor	223
Fe doped ZnO	PVA	Uniaxial – calcination	Regular	T_c > 300 K		410
Co doped ZnO	PVP	Uniaxial – calcination	Regular	Ferromagnetic, M_s = 0.05 emu g^{-1}, H_c = 53.2 Oe, magnetic loss factor = 0.170–0.223	EWA	411
Fe doped SnO_2/TiO_2	PVP	Uniaxial – calcination	Beaded fibres	Ferromagnetic, M_s = 0.02–0.37 emu g^{-1}	Photocatalyst	412
Mn doped SnO_2	PVP	Uniaxial – calcination	Hollow	Ferromagnetic, paramagnetic, H_c = 15 kOe		221
Cu doped SnO_2	PVP	Coaxial – calcination	Hollow	RT ferromagnetism		413
GO doped $CoFe_2O_4$	PVP	Uniaxial electrospinning – calcination	Regular	Ferrimagnetic, M_s = 79.24–82.7 emu g^{-1}, H_c = 909–1514 Oe, M_r = 31–39 emu g^{-1}		224
La-Doped $TiO_2/CoFe_2O_4$	PVP	(Sol–gel) – two-spinneret	Regular	M_s = 8.888 emu g^{-1}	Photocatalytic	414
Co-Doped $SrTiO_3$	PVP	Uniaxial – calcination – anneal	Regular	Paramagnetic–weak ferromagnetic		129
Gd doped bismuth ferrite	PVP	Uniaxial – annealing	Regular	M_s = 2.4–4.12 emu g^{-1}, H_c = 450 Oe		415
Ca doped $BiFeO_3$	PVP	Uniaxial – calcination	Sintered particles to form a fibre	Ferromagnetic	Photocatalyst	240
Bi_2O_3 doped $Ni_{0.5}Zn_{0.5}Fe_2O_4$	PVP	Uniaxial – calcination	Regular	M_s = 2.6–59.1 emu g^{-1}, H_c = 32.6–112.9 Oe		416
Fe_xCo_y	PAN/PBZ	Uniaxial – activation – carbonisation	Regular	M_s = 18.76 emu g^{-1}	Catalyst for PMS activation	417
	PVA	Uniaxial – graphinisation	Fibres encapsulated in graphite shell	Ferromagnetic, M_s = 71.14 emu g^{-1}, H_c = 220 Oe (300 K), H_c = 648 Oe (5 K)		225

Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
Fe-Ni	PVP	Uniaxial – calcination – deoxidation	Regular	Ferromagnetic, M_s = 72.54–195.06 emu g^{-1}, H_c = 1.72–43.89 Oe		128
		Uniaxial – calcination – deoxidation	Nanoribbons	Soft magnetic, M_s = 145.7 emu g^{-1} (max), H_c = 132 Oe		126
Fe$_3$Si	PVP	Uniaxial – calcination	Fibres with deposited particles	Ferromagnetic, M_s = 0–8.4 emu g^{-1}, H_c = 50–90 Oe	Tunable EM and microwave absorption	418
FePt	PVP	Uniaxial – calcination – reduction process	Necklace-like sintered particles to form a fibre	Hard magnetic, M_s = 54.63–59.38 emu g^{-1}, H_c = 4.68–10.27 Oe, M_r = 24.44–34.23 emu g^{-1}		123
CoNi	PVA	Uniaxial – calcination	Regular	Ferromagnetic, M_s = 47.45 emu g^{-1}, H_c = 65.6 Oe, M_r = 5.94 emu g^{-1}, H_s = 4000 Oe		419
CoPt	PVA	Uniaxial – carbonisation	Bead	Ferromagnetic, M_s = 77.3 emu g^{-1} (Co, max), H_c = 270.9 Oe (Co–Pd, max), M_r = 7.98 emu g^{-1} (Co, min)		420
Sm$_2$Co$_{17}$	PVP	Uniaxial – blend – calcination	Regular	M_s = 55.5–106 emu g^{-1}, M_r = 35.6–52.5 emu g^{-1}, H_c = 5210–12 676 Oe		206
Co$_x$Fe$_y$Al	PVP & PVA	Uniaxial – anneal	Regular	Ferromagnetic		421
SmCoFe	PVP	Uniaxial – calcination – REDOX post – treatment	Sintered particles to form a fibre	M_s = 80–120 emu g^{-1}, H_c = 5–7.5 kOe, M_r = 55–69 emu g^{-1}, M_r/M_s = 0.54–0.69		422
Co–MnO	PVA	Uniaxial – calcination	Regular	Ferromagnetic, M_s = 49.95 emu g^{-1}, H_c = 245 Oe, M_r = 7.35 emu g^{-1}, H_s = 3000 emu g^{-1}		423
Zn$_{1-x}$Co$_x$O	PVP	Uniaxial – calcination – anneal	Regular	RT ferromagnetic, M_s = 0.877 emu g^{-1}, H_c = 610 Oe (max)		424
	PVA	Uniaxial – sinter	Regular	H_c = 50–75 Oe		218
Zn–Mn–O	TPEE	Uniaxial – calcination	Microsphere composed of micro/nanofibres	Ferromagnetic, M_s = 0.20225–0.78425 emu g^{-1}, H_c = 83.68–223.78 Oe, M_r = 0.010125–0.015766 emu g^{-1}	Photocatalyst	425
Fe$_2$O$_3$, Fe$_3$O$_4$, CuFe$_2$O$_4$, Cu$_2$Fe$_2$O$_4$	PAN	Coaxial – calcination	Porous	Ferromagnetic, M_s = 2.058 emu g^{-1}, M_r = 0.28148 emu g^{-1}, H_c = 167.56 Oe	EWA	426
Magnetically susceptible conjugation complex		Uniaxial – deposition – surface modification	Regular		Biosensor	427
CoFe$_2$O$_4$	PVP	Uniaxial – anneal	Nanotubes	M_s = 18 emu g^{-1}	Photocatalysts	164
		Uniaxial – calcination – (in situ) oxidative polymerisation method	Hollow core–double shell nanostructure		Photocatalysts	158
		Uniaxial – calcination	Wrinkle nanofibres with cluster	M_s = 33.232 emu g^{-1}, H_c = 893.71 Oe, M_r = 10.876 emu g^{-1}	Photocatalyst	200
		Uniaxial – calcination	Nanorod with flake surface	Ferromagnetism, M_s = 35.17–61.24 emu g^{-1}	Photocatalyst	202
		Uniaxial – calcination – redox post – treatment	Regular		Efficient catalysts for the p-nitrophenol hydrogenation	428
		Uniaxial – calcination	Sintered particles to form a fibre	M_s = 53.2–71.7 emu g^{-1}, H_c = 925.3–1161.7 Oe		429
		Uniaxial – calcination	Sintered particles to form a fibre			106
		Uniaxial – anneal	Janus	M_s = 41.34 emu g^{-1}		75
		Uniaxial – calcination	Sintered particles to form a fibre	Soft magnetic, superparamagnetic, M_s = 28.61–67.24 emu g^{-1}, H_c = 758.62–2221.51 Oe, M_r = 8.28–35.11 emu g^{-1}		172
		Uniaxial – calcination	Sintered particles to form a fibre	Ferromagnetic, M_s = 42.8 emu g^{-1}, H_c (bulk) = 750–1000 Oe, M_r/M_s = 0.27–0.5		430
		(Dual-channel) – calcination	Sintered particles to form two phases that are distributed semi-cylindrically	H_c = 250 Oe		431
		Uniaxial – calcination	Nanotubes	H_c = 300 Oe (360 K), H_c = 10 400 Oe (5 K)		432

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Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
		Uniaxial – ultrasonic – coaxial	Regular	$M_s = 3.65$–45.80 emu g^{-1}, $H_c = 735$–785 Oe, $M_r = 1.4$–16.49 emu g^{-1}		433
		Uniaxial – calcination	Nanoribbons	ferromagnetic, $M_s = 64.6$–80.3 emu g^{-1}, $H_c = 1223$–1802 Oe		119
		Uniaxial – calcination	Nanoribbons (400–700 °C), nanofibres (900 °C)	$M_s = 76.2$–85.2 emu g^{-1} (2–300 K), $H_c = 895$ Oe (max), $M_r/M_s = 0.75$–0.89		114
		Uniaxial – calcination	Hollow	Ferromagnetic, $M_s = 9.0$–34.7 emu g^{-1}, $H_c = 858$–1207 Oe, $M_r = 2.7$–13.1 emu g^{-1}, $M_r/M_s = 0.30$–0.38 (300 K), $M_s = 9.6$–36.1 emu g^{-1}, $H_c = 11\,953$–14\,110 Oe, $M_s = 8.3$–32.5 emu g^{-1}, $M_r/M_s = 0.86$–0.89 (2 K)		160
		Uniaxial – calcination	Wire-in-tube structure	$H_c = 11\,043$ Oe (10 K), $H_c = 707$ Oe (300 K)		434
		Coaxial – anneal	Core-shell	$M_s = 16.1$ emu g^{-1}		156
		Uniaxial – coprecipitate – calcination	Nanofibres with deposited particles		Photocatalyst	242
		Uniaxial – anneal	Porous nanoribbons	$M_s = 56$ emu g^{-1}, $H_c = 757$ Oe, $M_r = 18$ emu g^{-1} (300 K), $M_s = 72$ emu g^{-1}, $H_c = 14\,507$ Oe, $M_r = 56$ emu g^{-1} (2 K)		115
	PAN	Uniaxial – carbonisation	Regular	$M_s = 30$ emu g^{-1}	Catalyst for PMs activation	243
		Uniaxial – stabilisation – carbonisation	Regular	$M_s = 50$–63 emu g^{-1}, $H_c = 0$–667 Oe, $M_r = 0$–17 emu g^{-1}		94
	PANI	Uniaxial – calcination – polyaniline assisted (self-assembly) process – deposition	Hollow fibres with deposited particles	$M_s = 35$ emu g^{-1}, $H_c = 1260$ Oe	Catalysts	244
	PVAc	(Sol–gel) uniaxial – calcination	Regular	$M_r = 16.1$–32.5 emu g^{-1}, $H_c = 611.9$–786.5 Oe		194
	PVDF	Uniaxial – blend – calcined – casted	Regular	$M_s = 10.7$ emu cm^{-3}		435
CoFe$_2$O$_4$/barium carbonate	PVP	Uniaxial – calcination – electrical assembly	Janus	Ferromagnetic, $M_s = 60$ emu g^{-1}, $H_c = 800$ Oe, $M_r = 18$ emu g^{-1}	Magnetoelectric sensors	436
CoFe$_2$O$_4$/Ag	PVP	Uniaxial – calcination	Sintered particles to form a hollow fibre		Catalyst for the degradation of organic pollutants	235
CoFe$_2$O$_4$–Pb(Zr$_{0.52}$Ti$_{0.48}$)O$_3$	PS	Uniaxial – anneal	Regular	Ferromagnetic, $H_c = 386$–730 Oe, $M_r = 3.3$–11.3 emu g^{-1}		437
	PVP & PMMA	Coaxial – anneal	Core-shell	Ferromagnetic, $H_c = 700$ Oe, $M_r = 3.40$ emu g^{-1}		140
CoFe$_2$O$_4$/CoFe$_2$	PVP	Uniaxial – calcination – partially reduction	Regular	Ferromagnetic, hard magnetic, $M_s = 66.8$–220.2 emu g^{-1}, $H_c = 0.62$–1.37 Oe, $M_r = 27.6$–106.4 emu g^{-1}		124
CoFe$_2$O$_4$/SrFe$_{12}$O$_{19}$	PVP	Uniaxial – calcination	Hollow	$M_s = 52.9$–62.8 emu g^{-1}, $H_c = 1089$–4046 Oe, $M_r = 18.22$–31.21 emu g^{-1}		438
CoFe$_2$O$_4$, NiFe$_2$O$_4$	PVP	Coaxial – calcination	Core-shell, fibre-in-tube, tube-in-tube	$M_s = 63.83$–76.16 emu g^{-1}, $H_c = 12.79$–13.36 kOe		439
ZnFe$_2$O$_4$/CoFe$_2$O$_4$	PVP	Uniaxial – calcination	Short	$M_s = 53.95$–69.62 emu g^{-1}, $H_c = 69$–110 Oe, $M_r = 3.8$–11 emu g^{-1}, $M_r/M_s = 0.07043$–0.15799		440
Mullite fibres with embedded Ni NPs	AN & PEO	Uniaxial – thermal reduction – heat treatment	Regular	Ferromagnetic, $M_s = 0.085$–4.177 emu g^{-1}, $H_c = 21.1$–85.8 Oe, $M_r = 0.005$–1.136 emu g^{-1}		132
Fe–Ni/NiFe$_2$O$_4$	PVP	Uniaxial – calcination – partially reduction	Regular	Soft magnetic, $M_s = 49.5$–109.3 emu g^{-1}, $H_c = 219$–460 Oe		136
NiFe$_2$O$_4$ & MWCNTs	PAN	Uniaxial – stabilisation – carbonisation	Regular	Ferromagnetic, $M_s = 1.5$ emu g^{-1}, $H_c = 47$ Oe	EM shielding	142
Spinel-NiMn$_2$O$_4$	PVP	Uniaxial – anneal	Short fibres	Paramagnetic-antiferromagnetic (200–300 K), $M_s = 34$ emu g^{-1}, $H_c = 96.7$ Oe, $M_r = 13$ emu g^{-1} (5 K)	Electrode material and energy storage	175
NiFe$_2$O$_4$	PVP	Uniaxial – calcination – atomic layer deposition	Core-shell		Photocatalyst	441

Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
		Uniaxial – calcination	Multi-particle-chain-like	$M_s = 2.7 \times 10^8$ emu cm^{-3}, $H_c = 166$ Oe, $T_c = 858$ K		442
		Uniaxial – calcination	Sintered particles to form a fibre	Soft ferromagnetic, $M_s = 1.3$–40 emu g^{-1}, $H_c = 16$–170 Oe, $M_r = 0.008$–7.8 emu g^{-1}		183
		Uniaxial – calcination	Regular	$H_c = 30$ Oe (300 K), $H_c = 576$ Oe (5 K)		443
	PVA	Uniaxial – anneal	Nanotube	Ferromagnetic, $M_s = 33.3$ emu g^{-1}, $H_c = 225$ Oe		444
	PVA & TEOS	Uniaxial – anneal	Regular	Ferromagnetic, $H_c = 60$ Oe		445
		Uniaxial – (dip-coating) – calcination	Hierarchical porous cross-linked structure	Soft magnetic, $M_s = 14.44$ emu g^{-1}		110
NiCo$_2$O$_4$	PAN	Uniaxial – stabilisation – carbonisation	Regular		EWA	229
Ni$_{0.8}$Co$_{0.2}$Fe$_2$O$_4$, Ni	PVP, PAN	Uniaxial – carbonisation – mix with paraffin	Regular	$M_s = 11.9$–53.4 emu g^{-1}, $H_c = 114.3$–409 Oe	EWA	446
Co$_{0.5}$Ni$_{0.5}$Fe$_2$O$_4$	PVP	Uniaxial – calcination	Necklace-like	$M_s = 37.5$–66.2 emu g^{-1}, $H_c = 78.3$ Oe (max)	EWA	122
Co$_{1-x}$Ni$_x$Fe$_2$O$_4$ ($x = 0.0, 0.2, 0.4, 0.6, 0.8, 1.0$)	PVP	Uniaxial – stabilisation – calcination	Regular	$M_s = 13.4$–56.4 emu g^{-1}, $H_c = 210$–1255 Oe		215
MnFe$_2$O$_4$	PVP	Uniaxial – calcination	Short fibres	$M_s = 43.5$–46.3 emu g^{-1}	Catalysts (magnetic separation)	447
		Uniaxial – calcination	Nanorods	$M = 46.8$–61.14 emu g^{-1} (10 kOe), $H_c = 607.7$–642.6 Oe, $M_r = 13.4$–17.8 emu g^{-1} (600–800 K)		174
	PVAc	Uniaxial – calcination	Regular		Magnetic recording device, magneto-optical recording and electronic devices	186
	PAN	Uniaxial – carbonisation	Regular	$M_s = 56.8$ emu g^{-1} (max), $H_c = 998$–1264 Oe, $M_r = 15$–31.9 emu g^{-1}, $M_r/M_s = 0.41$–0.67		185
Co$_{0.5}$Mn$_{0.5}$Fe$_2$O$_4$	PVP	Uniaxial – calcination – ANI polymerisation	Hollow		Photocatalyst	448
CuFe$_2$O$_4$	PVP	Uniaxial – calcination	Hollow	Ferromagnetic, $M_s = 18.99$–25.04 emu g^{-1}, $H_c = 345$–1451 Oe, $M_r = 5.32$–13.02 emu g^{-1}, $M_r/M_s = 0.28$–0.52		159
		Uniaxial – deposition	Nano fibres with deposited particles		Visible light photocatalyst (magnetic separation)	199
		Uniaxial – calcination	Sintered particles to form a fibre or lamellar post sintering	Ferromagnetic, soft magnetic, $M_s = 7.73$–23.98 emu g^{-1}, $H_c = 299$–625 Oe, $M_r = 6.08$–9.91 emu g^{-1}, $M_r/M_s = 0.34$–0.41		182
		Uniaxial – calcination	Sintered particles to form short or hollow fibres	Ferromagnetic	CR adsorption and CO catalytic oxidation of the samples	449
CuCo$_2$O$_4$	PAN	Uniaxial – calcination	Regular	Ferromagnetism – antiferromagnetism, $M_s = 11.43$–60.24 ($\times10^{-3}$) emu g^{-1}, $H_c = 107.07$–734.67 Oe, $M_r = 4.02$–25.78 ($\times10^{-3}$) emu g^{-1}, $M_r/M_s = 0.289$–0.428		190
Ni$_{1-x}$Cu$_x$Fe$_2$O$_4$	PVP	Uniaxial – calcination	Regular	Soft ferromagnetic, $M_s = 15.1$–37.71 emu g^{-1}, $H_c = 50.9$–144.66 Oe		210
ZnFe$_2$O$_4$	PVP	Uniaxial – anneal	Porous nanotubes	Ferromagnetic, $M_s = 12.4$ emu g^{-1}, $H_c = 48.79$ Oe	Photocatalyst	450
	PVP	Uniaxial – calcination	Sintered particles to form a fibre		EWA	451
		Uniaxial – calcination	Sintered particles to form a fibre	Superparamagnetic – paramagnetic (calcined at 500–700 °C), $M_s = 1.53$–2.55 emu g^{-1}		193
ZnFe$_2$O$_4$/ZnO	PVP	Uniaxial – calcination	Porous nanotubes		Photocatalyst	452
ZnFe$_2$O$_4$/Fe$_3$O$_4$	PVP	Uniaxial – calcination	Regular	Ferromagnetic, $M_s = 22.9$–34.2 emu g^{-1}, $H_c = 208.7$–425.1 Oe, $M_r = 5.8$–10.1 emu g^{-1}	Photocatalysts for wastewater disposal or drug release	453

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Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
$ZnFe_2O_4/\gamma\text{-}Fe_2O_3$	PVA	Uniaxial – calcination	Nanoribbons	Ferromagnetic, $M_s = 45$ emu g^{-1} (max)		454
$Cu_{1-x}Zn_xFe_2O_4$	PVP	Uniaxial – calcination	Regular	Ferromagnetic-paramagnetic, $M_s = 58.4$ emu g^{-1} (max), $H_c = 35.2\text{–}723.5$ Oe, $M_r/M_s = 0.11\text{–}0.47$ emu g^{-1}		216
$Co_{0.4}Zn_{0.4}Fe_2O_4$	PVP	Uniaxial – calcination	Regular	Ferromagnetic, $M_s = 92.3$ emu g^{-1} (max), $H_c = 338.2$ Oe (max)	EWA	455
$Ni_{0.5-x}Cu_xZn_{0.5}Fe_2O_4$	PVP	Uniaxial – calcination	Regular	$H_c = 121.6$ Oe (298 K), $H_c = 295.9$ Oe (77 K)		211
$Ni_{0.5}Zn_{0.5}Fe_2O_4$	PVP	Uniaxial – anneal – calcination	Nanowires in nanotubes	$M_s = 61$ emu g^{-1}, $H_c = 43$ Oe, $M_r = 3$ emu g^{-1}		456
NiZn ferrite	PVP	Uniaxial – calcination	Regular	Soft magnetic, $M_s = 78.6$ emu g^{-1}, $H_c = 57.4$ Oe	EWA	457
	PVP	Uniaxial – calcination	Sintered particles to form a fibre	Ferromagnetic	DNA separation for clinical diagnoses and biomolecular recognition	458
$MgFe_2O_4$	PVP	Uniaxial – calcination	Sintered particles to form a fibre	$M_s = 17\text{–}31.1$ emu g^{-1}, $H_c = 35.8\text{–}98.9$ Oe		184
$Mg_{1-x}Zn_xFe_2O_4$	PVP	Uniaxial – anneal – sinter	Nanotubes	Ferromagnetic		135
	PVP	Uniaxial – calcination	Regular	$M_s = 20.25$ emu g^{-1}, $M_r = 5.1$ emu g^{-1}, $H_c = 90$ Oe, soft ferromagnetic		459
$CuAl_{0.95}Co_{0.05}O_2$	PVP	Uniaxial – anneal	Regular	Ferromagnetic, $M_s = 0.012$ emu g^{-1}, $H_c = 65.26$ Oe, $M_r = 0.0015$ emu g^{-1}		104
Fe@TiSi	P123, PVP	Coaxial – calcination	Core-shell	$M_s = 0.0002\text{–}1.2$ emu g^{-1}	Catalyst for waste-water treatment	249
$\alpha\text{-}Fe_2O_3@SiO_2$	TEOS	Uniaxial – calcination – deposition	Core-shell	$M_s = 14\text{–}20$ emu g^{-1}, $H_c = 390\text{–}400$ Oe		151
$SiO_2\text{–}CoFe_2O_4$	PVP & TEOS	Uniaxial – anneal	Hollow short fibres	Ferromagnetic, $M_s = 56.4\text{–}80$ emu g^{-1}, $H_c = 1477$ Oe		201
$CaFe_2O_4$	PVP	Uniaxial – calcination	Necklace-like	Superparamagnetic	Photocatalyst	238
$CaFe_2O_4/MgFe_2O_4$	PVP	Uniaxial – calcination	Sintered particles to form fibres		Photocatalyst	460
$TiO_2/CoFe_2O_4$	PVP	(Sol–gel) – vertical two – spinneret – calcination	Regular	$H_c = 585.09$ Oe, $M_r = 3.2408$ emu g^{-1}, $M_s = 9.5869$ emu g^{-1}	Photocatalyst	461
$Ti_{0.9}V_{0.1}O_2$	PVP	Uniaxial – calcination	Regular	Ferromagnetic		191
$Cr_{0.046}Zn_{0.954}O$	PVP	Uniaxial – calcination	Regular	Weak ferrimagnetic, $H_c = 73\text{–}224$ Oe		462
$SrFe_{12}O_{19}$	PVP	Coaxial – stabilisation – sinter	Hollow	$M_s = 30.7\text{–}57.8$ emu g^{-1}, $H_c = 95.3\text{–}4187.1$ Oe	EWA	167
		Uniaxial – calcination – deposition – calcination	Core-sheath	Ferromagnetic, $M_s = 7.968$ emu g^{-1}, $H_c = 5149.7$ Oe, $M_r = 4.235$ emu g^{-1}	Photocatalyst	3
		Uniaxial – calcination	Necklace-like	$M_s = 64$ emu g^{-1}, $H_c = 6533.3$ Oe	High density magnetic recording and microwave devices	138
		Uniaxial – anneal	Nanoribbons	$M_s = 50\text{–}67.9$ emu g^{-1}, $M_r = 27.5\text{–}37.3$ emu g^{-1}, $H_c = 7150\text{–}7310$ Oe		116
$SrFe_{12}O_{19}/FeCo$	PVP	Uniaxial – calcination	Necklace-like	$M_s = 59.9\text{–}60.8$ emu g^{-1}, $H_c = 4538.2\text{–}6565.9$ Oe		176
$SrAl_xFe_{12-x}O_{19}$ (x = 0–3.0)	PVP	Coaxial – calcination – reduction	Core-shell	$M_s = 60.9\text{–}68.8$ emu g^{-1}, $H_c = 1249\text{–}3190$ Oe		153
$SrFe_{12}O_{19}$	PVP	Uniaxial – calcination	Necklace-like	$M_s = 13\text{–}62$ emu g^{-1}, $H_c = 5668\text{–}7737$ Oe (298 K), $M_s = 16\text{–}85$ emu g^{-1}, $H_c = 6860$ Oe (77 K)		209
$SrTi_{1-x}Co_xO_3$	PVP	Uniaxial – calcination – anneal	Regular	Ferromagnetic, $M_s = 0.74$ emu g^{-1} (max)		130
$SrTi_{1-x}Fe_xO_3$ ($SrTi_{0.9}Fe_{0.1}O_3$)	PVP	Uniaxial – calcination	Regular	Paramagnetic-ferromagnetic, $M = 0.46\text{–}0.82$ emu g^{-1}, $H_c = 165\text{–}217$ Oe, $M_r = 0.01\text{–}0.15$ emu g^{-1}, $T_c > 300$ K	Batteries	463
$SrTiO_3/SrFe_{12}O_{19}$	PVP	Uniaxial – calcination	Regular	Hard magnetic, $M_s = 55.1$ emu g^{-1}, $H_c = 6523$ Oe (max), $M_r = 5.94$ emu g^{-1}		181
$SrFe_{12}O_{19}$, $Ni_{0.5}Zn_{0.5}Fe_2O_4$	PVP	Uniaxial – calcination	Sintered particles to form a fibre	$H_c = 3037.1\text{–}4762.7$ Oe (77–297 K), $M_r = 53.5\text{–}39.1$ emu g^{-1} (77–297 K)		464

Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
		Uniaxial – calcination	Sintered particles to form a fibre	M_s = 56.1–64.9 emu g^{-1}, H_c = 1484.7 Oe, M_r = 31.5 emu g^{-1}		465
xSrSiO$_3$/(100 − x)SrFe$_{12}$O$_{19}$	PVP	Uniaxial – calcination	Necklace-like	M_s = 45.6–58.0 emu g^{-1}, H_c = 6283.8 Oe (max)		466
Yttrium iron garnet	PVP	Uniaxial – presintering – calcination	Sintered particles to form a fibre	M_s = 3.1–21.5 emu g^{-1}, H_c = 21.5–140 Oe		170
BaFe$_{12}$O$_{19}$	PVP	Coaxial – calcination	Hollow	M_s = 46.15–51.56 emu g^{-1}, H_c = 5226 Oe, M_r = 22.54–27.59 emu g^{-1}		467
		Uniaxial – calcination	Sintered particles to form a hollow fibre	M_s = 17.8 emu g^{-1}, H_c = 4106.9 Oe		468
		Uniaxial – calcination	Sintered particles to form a fibre	M_s = 71.5 emu g^{-1}, H_c = 5943 Oe		469
BaFe$_{12-x}$Al$_x$O$_{19}$	PVP	Uniaxial – calcination	Sintered particles to form a fibre	Hard magnetic, M_s = 29.70–63.92 emu g^{-1}, H_c = 3614.0–9288.4 Oe, M_r = 17.73–33.44 emu g^{-1}, M_r/M_s = 0.52–0.60		212
BaTi$_{0.90}$Mn$_{0.10}$O$_3$	PVP	Uniaxial – anneal	Sintered particles to form a fibre	Paramagnetic		222
Ni$_{0.4}$Co$_{0.2}$Zn$_{0.4}$Fe$_2$O$_4$/BaTiO$_3$	PVP	Uniaxial – stabilisation – calcination	Necklace-like	M_s = 34.5–64.4 emu g^{-1}, H_c = 93.2–132.8 Oe	EWA	470
Fe$_2$O$_3$, BaTiO$_3$	PAN	Uniaxial – stabilisation – carbonisation	Regular	Superparamagnetic, M_s = 12–12.5 emu g^{-1}, H_c = 87–95 Oe	Electromagnetic interference shields	471
Ba$_{0.7}$Sr$_{0.3}$TiO$_3$–Ni$_{0.8}$Zn$_{0.2}$Fe$_2$O$_4$	PVP	Uniaxial – anneal	Regular	Ferromagnetic, M_s = 2.479 emu g^{-1}, H_c = 44.873 Oe, M_r = 0.189 emu g^{-1}		472
Cobalt ferrite/barium calcium titanate	PVP	Uniaxial – magnetic field – sinter	Regular	Soft magnetic		473
Nickel ferrite/barium titanate	PVP	Coaxial – anneal	Core–shell	Ferromagnetic, H_c = 28 078 Oe, M_r = 0.23 emu g^{-1}		48
LaFeO$_3$	PVP	Uniaxial – calcination	Short fibres	T_c = 255 K, T_B = 180 K		171
LaMnO$_{3+\delta}$	PVA	Uniaxial – calcination	Null	Ferromagnetic, T_c ≈ 365 K		474
La$_{1-x}$Sr$_x$MnO$_3$	PVP	Uniaxial – calcination	Regular	Ferromagnetic, M_s = 0.022–0.067 emu g^{-1}		475
La$_{0.5}$Sr$_{0.5}$TiO$_3$	PVP	Uniaxial – calcination	Regular	M_s = 1.23–40.52 emu g^{-1}, H_c = 62 Oe (max)		476
La$_{0.7}$Sr$_{0.3}$MnO$_3$	PVP	Uniaxial – calcination	Regular			179
	PVP	Uniaxial – calcination	Hollow	Superparamagnetic–ferromagnetic, M_s = 1.8–50 emu g^{-1} (max)		180
Ba$_{1-x}$La$_x$Fe$_{12}$O$_{19}$	PVP	Uniaxial – calcination	Regular	M_s = 77.188 emu g^{-1}, H_c = 3559.0 Oe (min)	EWA	208
Sr$_{0.8}$La$_{0.2}$Zn$_{0.2}$Fe$_{1.8}$O$_{19}$	PVP	Uniaxial – calcination	Necklace-like	M_s = 43.2–72.2 emu g^{-1}, H_c = 164.3–5478.7 Oe		477
Sr$_{1-x}$La$_x$Fe$_{12-x}$Co$_x$O$_{19}$ (x = 0.12)	PVP	Uniaxial – sinter	Regular	M_s = 70.76 emu g^{-1}, H_c = 6.26 kOe, M_r = 36.35 emu g^{-1}		478
Ce$_{0.96}$Fe$_{0.04}$O$_2$	PVP	Uniaxial – calcination	Null	Ferromagnetic, T_c > 390 K, M_s = 0.0025–0.0923 emu g^{-1}, H_c = 10 kOe		220
CuFe$_2$O$_4$@CeO$_2$	PVP	Uniaxial – calcination – precipitation – calcination	Core–shell	M_s = 20.51–28.32 emu g^{-1}, M_r = 7.24–12.85 emu g^{-1}	Dye removal	152
SnO$_2$/Ce	PVP	Uniaxial – calcination	Porous hollow	Ferromagnetic, M_s = 19 × 10^{-5} emu g^{-1}		479
La$_{0.33}$Pr$_{0.34}$Ca$_{0.33}$MnO$_3$	PVP	Uniaxial – calcination	Regular	T_c ≈ 150 K, T_B ≈ 50 K	Magnetoresistance	219
Hard Sm$_2$Co$_{17}$ core and soft Fe$_{11}$Co$_5$ shell	PVP	Uniaxial – calcination – soft magnet plating	Core–shell		Permanent magnet	154
α-NaYF$_4$:Yb/Er/Gd	PEI	Uniaxial – anneal	Hollow		Drug delivery	165
Na(Y/Gd)F$_4$:Yb^{3+},Er^{3+}	PVP	Uniaxial – anneal	Porous		Drug delivery	480
GdOF:Er^{3+}	PVP	Uniaxial – calcination – fluorination – oxidation	Regular	Paramagnetic, mass magnetic susceptibility = 1.4252 × 10^{-4} emu g^{-1} Oe^{-1} (±20 Oe)		481
GdF$_3$:Eu^{3+}	PVP	Uniaxial – calcination – fluorination	Regular	Paramagnetic, magnetisation, M_s = 2.11–2.46 emu g^{-1}		482
NaGdF$_4$:Dy^{3+}	PVP	Uniaxial – calcination – fluorination	Nanobelts	Superparamagnetic, M_s = 34.17–56.13 emu g^{-1}, H_c = −50 to 50 kOe, mass magnetic susceptibility = 2.2239–3.7438 × 10^{-5} emu g^{-1} Oe^{-1}		125

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Table 2 (continued)

Magnetic ingredients	Polymer	Electrospinning approach and post-treatments	Morphology (after post-treatment)	Magnetic properties	Applications	Ref.
$NaGdF_4;0.5\%Dy^{3+},Eu^{3+}$	PVP	Uniaxial – calcination – fluorination	Regular	Magnetisation = 1.69–2.00 emu g^{-1}		483
$Gd_2O_2S:Dy^{3+},Eu^{3+}$	PVP	Uniaxial – sinter – sulfurising	Regular	M_s = 2.56–2.68 emu g^{-1}		484
$SrDy_xFe_{12-x}O_{19}$	PVP	Uniaxial – calcination	Regular	Hard magnetic, M_s = 53–64 emu g^{-1}, H_c = 6835–7155 Oe		155
$Ni_{0.8}Gd_{0.2}Fe_2O_4$	PVP	Uniaxial – calcination	Nanoribbon	M_s = 29.45–36.71 emu g^{-1}, H_c = 8.7–33.73 Oe, M_r = 0.52–1.93 emu g^{-1}		213
$Co_{0.5}Cu_{0.2}Ni_{0.1}Fe_2O_4$, $Co_{0.94}Gd_{0.06}Fe_2O_{4.03}$	PVP	Uniaxial – calcination	Sintered particles to form a fibre	Soft magnetic, M_s = 33.32–67.37 emu g^{-1} ($Co_{0.5}Cu_{0.2}Ni_{0.2}Zn_{0.1}Fe_2O_4$), M_s = 12.44–71.98 emu g^{-1} ($Co_{0.94}Gd_{0.06}Fe_2O_{4.03}$)		214
$BiFeO_3$	PVP	Uniaxial – calcination	Regular	Weak ferromagnetic	Photocatalyst	45
		Uniaxial – stabilisation – anneal	Regular	Ferromagnetic, good magnetic response, M_s = 4.4 emu g^{-1}, H_c = 170 Oe	Photocatalyst	239
		Uniaxial – calcination – hydro-thermal method	Nanofibres with nanosheet deposited	Ferromagnetic	Photocatalyst	485
		Uniaxial – anneal	Short	Weak-ferromagnetic		139
		Uniaxial – calcination	Necklace-like	Magnetic moment = 0.4–3.0 emu g^{-1}		486
		Uniaxial – anneal	Nanotubes	Ferromagnetic, T_c = 300 K		113
		Uniaxial – anneal	Regular	Weak ferromagnetic, M_s = 4 emu g^{-1} (max), H_c = 200 Oe		487
		Uniaxial – calcination	Regular	Ferromagnetic, M_s = 4 emu g^{-1}		207
		Uniaxial – calcination – grind – Uniaxial	Regular	Ferromagnetic, M_s = 0.18–0.36 emu g^{-1}		488
$Ni_{0.5}Zn_{0.5}Fe_2O_4/Pb(Zr_{0.52}Ti_{0.48})O_3$	Nylon-6	Uniaxial – calcination	Regular	Ferromagnetic		489
$Bi_5Ti_3FeO_{15}$	PVP	Uniaxial – calcination	Regular	Ferroelectric, soft magnetic		178
	PVP	Uniaxial – calcination	Compact pellet to form a fibre	Weak ferromagnetic, M_s = 6.7 × 10^{-4} emu g^{-1}, H_c = 255 Oe, M_r = 0.92 × 10^{-4} emu g^{-1}, M_r/M_s = 0.14		490
$Bi_2Sr_2CaCu_2O_{8+x}$	PVP	Uniaxial – anneal – calcination	Regular	T_c = 78.7 K		491
$Bi_{0.95}Y_{0.05}FeO_3$	PVP	Sol–gel uniaxial – calcination	Sintered particles to form a fibre	M = 1.143–1.996 emu g^{-1} (MH = 2 T)		492
$Bi_{0.9}La_{0.1}Fe_{0.95}Mn_{0.05}O_3$	PVP	Uniaxial – calcination	Regular	M_s = 404.7 memu g^{-1}, H_c = 687.72 Oe, M_r = 48.115 memu g^{-1}		493
$Li_7La_3Zr_2O_{12}$	PVP	Uniaxial – calcination	Regular			494
$Nd_{0.1}Bi_{0.9}FeO_3$	PVP	Sol–gel – uniaxial – carbonisation	Hollow	Antiferromagnetic, M_s = 1.82 emu g^{-1}, H_c = 300 Oe		495
$PbZr_{0.52}Ti_{0.48}O_3–NiFe_2O_4$	PVP	Coaxial – anneal – (self-assembly)	Core–shell	Magnetisation = 6.5–9.6 emu g^{-1}		157
$Fe_3O_4@SiO_2$	PVP	Uniaxial – blend – heated – hydrothermal treatment	Fibres with deposited particles	Superparamagnetic	Photocatalyst	496
Montmorillonite	PVDF	Uniaxial – blend	Agglomerates		EWA	497

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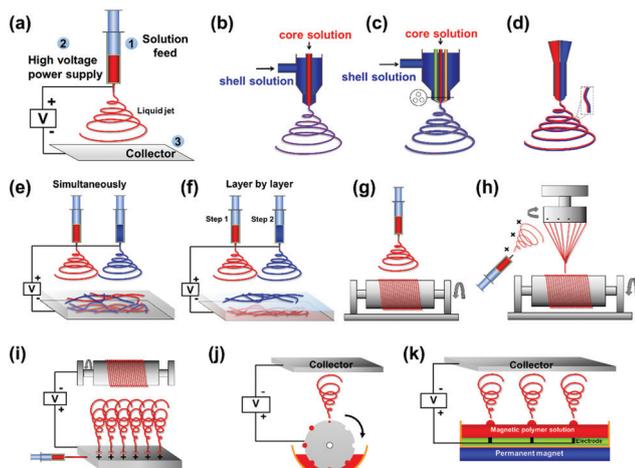
Fig. 2 Variety of different electrospinning apparatus including: (a) three basic parts, (b) core–shell needle, (c) multi-channel needle, (d) Janus fibre needle, (e) fibre blend, (f) layer-by-layer fibre, (g) drum collector, (h) twisted fibre collector, (i) multi-needle electrospinning, (j) drum electrospinning and (k) magnetically assisted electrospinning.
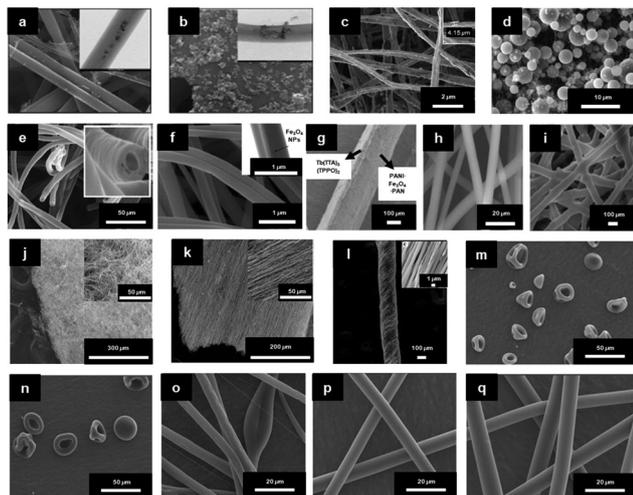
Fig. 3 SEM images of electrospun fibres and particles: (a) NFs with encapsulated MNPs;[8] (b) NFs with dip-coated MNPs;[8] (c) nanobelts;[55] (d) beaded fibres;[19] (e) hollow fibres;[26] (f) Janus NFs;[14] (g) cross section of a bi-layered composite nanofibrous film;[36] (h) the fibre of left layer containing the terbium complex Tb(TTA)$_3$(TPPO)$_2$ (where TTA is thenoyltrifluoroacetone and TPPO is triphenylphosphine oxide);[36] (i) the fibre of right layer containing PANI·Fe$_3$O$_4$·PAN;[36] reproduced from ref. 36 with permission from the PCCP Owner Societies; (j) random fibre mat;[45] (k) oriented fibre mat;[45] reproduced from ref. 45 with permission from the PCCP Owner Societies (l) yarn twist fibres[56] (Published by The Royal Society of Chemistry); and (m–q) electrosprayed and electrospun fibres of styrene–(ethylene-co-butylene)–styrene from neat tetrahydrofuran (THF) solutions at varying polymer concentration (8 wt%, 10 wt%, 12 wt%, 14 wt% and 18 wt% for m–q, respectively).[9]

differing diameter, which are loaded with the shell and core materials, respectively. The high speed of the jets prevents the disparate materials from mixing, resulting in a distinct boundary between the core and shell materials. Coaxial electrospinning is a useful method for separating the material of the inner and outer

layers whilst protecting the load material (typically contained within the core). Yu et al.[26] loaded air into the inner syringe to prepare hollow NFs (Fig. 3(e)) to be applied in targeted drug delivery applications.

2.1.2.3 Janus structure electrospinning. Janus nanostructures consist of two segregated materials with distinct physical and chemical properties to create a single nanostructure with two 'faces'. Commonly, the two materials exhibit antagonistic properties, such as being hydrophilic/hydrophobic (polar/nonpolar), which forms an important area of research in materials science. To create Janus NFs, two different fibres are produced from separate nozzles that have opposite charges. The two different fibres attract one another to form the final Janus structure (Fig. 3(f)). The Janus structure stops the two parts of the material interfering with one another to prevent the loss of dual performance. This technology has been utilised to produce materials with superior properties in catalysis, sensing, biomedicine and display technology.[27–30]

2.1.2.4 Collecting methods and collectors. In addition to nozzle design, and the whip and curing stages of the electrospinning process, the overall morphology of the fibre can be controlled by changing the method of collection. In addition to a stationary plate, common collection methods include (but are not limited to) additional magnetic field,[13,31,32] mechanical traction,[33] spinning[34] and layer-by-layer blending[35,36] (Fig. 2). For example, Yang et al.[35] prepared a novel sandwich-structured pellicle *via* layer-by-layer blending where the electric, magnetic and luminous layers could be effectively isolated from each other.

The arrangement of fibres can also be controlled by using different collectors. Again, compared to the more traditional flat plate, other collectors include the drum[37–41] (Fig. 2(g)), parallel roller[11,42–44] (Fig. 2(h)) and slit collector. These collectors allow NF membranes to be obtained with intricate patterns (Fig. 3(j–l)) for more innovative applications.[38,45] Among these collectors, the drum is the most common for preparing oriented fibre fabrics and membranes.[46]

2.1.2.5 Combinations. The aforementioned methods (discussed in Sections 2.1.2.1–2.1.2.4) are often combined to achieve enhanced control over the process. Such combinations have been exploited to access morphologies such as the yarn twist (Fig. 2(i)), Janus array fibre films[28,35,47] and oriented coaxial fibres.[48] Wang et al.[36] fabricated highly fluorescent membranes made up of Janus NFs composed of magnetic [Fe$_3$O$_4$/polyvinylpyrrolidone (PVP)] and fluorescent terbium ligand complex Tb(BA)$_3$phen/PVP (where BA is benzoic acid and phen is phenanthroline) NFs. The trifunctional bi-layered composite nanofibrous film was produced using layer-by-layer electrospinning and by systematically altering the process parameters during the electrospinning process.

2.1.2.6 Industrial production. In order to improve production efficiency, and to allow scale-up from the laboratory to industrial manufacture, modifications to the electrospinning equipment are often made. Most commonly the liquid supply device is adjusted commonly to; the use of densely packed needle arrays[49] (Fig. 2(j)), separate silk threads used as spinnerets,[50] a drum with grooves

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that rotates in a liquid pool to fill the hole for the supply of liquid (Fig. 2(k)) or a feedstock pool mounted with columns of magnets to directly supply the magnetic liquid[51] (Fig. 2(l)).

Generally, the development of electrospinning rigs has been based on the design and combination of the two main components: the solution feeding system and collector. The development of industrial mass production equipment has also contributed to the commercialisation of electrospinning fibres.

2.1.3 Morphology and parameters. The processing parameters that affect the structure and morphology of an electrospun fibre include the applied voltage, liquid flow rate and the distance between nozzle and collector. The applied voltage determines the electric field intensity and charge density of the droplet, which directly influence the diameter of the fibres produced. Generally, a simple increase in voltage, without changing any other parameters, results in a decrease in fibre diameter. However, special cases still exist. For example, beaded fibres can be produced at low polymer solution concentration and at increasing (but low) voltages, the diameter initially increases with decreasing bead density. It is only at significantly higher voltages that a reversal is observed; the fibre diameter decreases and the beads reappear.[52] Additionally, a voltage increase can also decrease the uniformity and length of the fibres produced.[53] Conversely, increasing the flow rate of the liquid results in an increase in fibre diameter. Sufficient time is required for the solvent to fully evaporate during the fibre forming process, which corresponds directly to the distance between the nozzle and collector. If the distance between the nozzle and collector is too short there is insufficient time for the solvent to evaporate and fused fibres are formed. Similarly, fused fibres are formed when the distance is too large, this is attributed to the reduced electrostatic field strength experienced by the fibres meaning they are not stretched appropriately. In this latter case, the fibres formed often have large diameters which is a result of solvent being trapped within.[54]

System properties such as polymer molecular weight, concentration, solution viscosity, solvent type and solution electroconductivity also affect the fibre morphology. Electrospinning relies on chain entanglement to produce fibres. The level of chain entanglement is directly related to the solution viscosity, which is intrinsically linked to the polymer molecular weight and sufficiently high solution concentration. Insufficient chain entanglement causes bead-like morphologies instead of continuous fibres. Wang et al.[9] studied the effect of co-solvent and polymer concentration on fibre morphology and the phenomenon of microphase separation during solution fibrillation (Fig. 3(m–q)). They found a morphological transition from fibres to beads occurred when increasing the concentration of dimethylformamide (DMF) in the THF/DMF co-solvent system. However, for the polymer system to self-assemble (microphase separate) the quantity of THF present had to be between 65–90 wt%. In another example, Doepke et al.[50] investigated nanoparticle concentration when preparing polymer bead/fibre mats for data storage. In this case they found that mechanical dispersion by ultrasonic treatment allowed for higher quantities of nanoparticles to be incorporated in both mats and bead formation without unwanted agglomeration effects.

MNPs or their constituent components (e.g. inorganic metal salts, alloys and oxides) are typically added into solution. These materials are highly electroconductive and alter the solution permittivity and conductivity, which in turn affects the creation of the local electrical field, improves the fibre morphology and decreases the fibre diameter.[52] Additionally, environmental factors such as humidity and temperature can also influence the fibre structure. Typically, these conditions relate to the speed of solvent volatilisation and thus affect the overall fibre morphology.

In order to obtain the desired morphology or functionality for the target application, the electrospun fibrous membrane is often post treated. The surface of the NFs can be coated with functional entities (e.g. collagen[57]) or heat treated to chemically crosslink the polymer to improve mechanical strength and/or prevent dissolution (e.g. for temperature-controlled drug release).[1] Sandwich structure fibrous membranes have also been prepared by thermally treating the electrospun sample post-deposition.[58] The magnetic mat is sandwiched between two non-magnetic mats before an alternating magnetic field is used to induce magnetic heating, which in turn thermally bonds the nanofibrous mats together.

Overall, the development of solution supply systems has provided a rich and varied internal structure of individual fibres and the development of the collector has resulted in a diverse range of inter-fibre structures.

2.2 Magnetic materials and properties

Magnetic nanomaterials such as Fe_3O_4, α/γ-Fe_2O_3 (hematite/maghemite) and MFe_2O_4 (M = metal) have received considerable attention due to their properties and potential applications. Herein, we review the magnetic materials that have been used in electrospun polymer-based NFs and how this relates to their final structure and properties.

2.2.1 Fe_3O_4. Fe_3O_4 (magnetite) is one of the most commonly used magnetic materials due to its ease of preparation, biocompatibility, high surface area, catalytic activity, electrical conductivity, low toxicity and almost full spin polarisation at room temperature (RT). Superparamagnetic[38,59,60] and ferrimagnetic[58,61] Fe_3O_4 NPs have been prepared with varying saturation magnetisation (M_s) values. The surface of Fe_3O_4 nanoparticles can also be modified to alter or enhance given properties. Song et al.[21] prepared Fe_3O_4–polyhedral oligomeric silsesquioxane (POSS) particles using a hydrosilylation reaction whilst maintaining an M_s value of the Fe_3O_4–POSS at 18.77 emu g^{-1}. The POSS was used in the system to improve the stability of the surface potential and charge retention of the mats. Gong et al.[62] synthesised Fe_3O_4-loaded multi-walled carbon nanotubes (MWCNTs) (as a template for the Fe_3O_4 NPs), achieving M_s as high as 71.549 emu g^{-1}. Fe_3O_4 NPs can also be mixed with a variety of polymer materials to increase their stability, introduce stimuli-responsiveness and manipulate the interparticle distance and magnetic interactions (Fig. 4). Additionally, polymers can guide the assembly of Fe_3O_4 NPs to form novel structures.[14,33,63–70]

Since polymeric materials can encapsulate and bind the nanoparticles as a matrix, research groups mix polymer materials with Fe_3O_4 NPs by uniaxial electrospinning, coaxial electrospinning
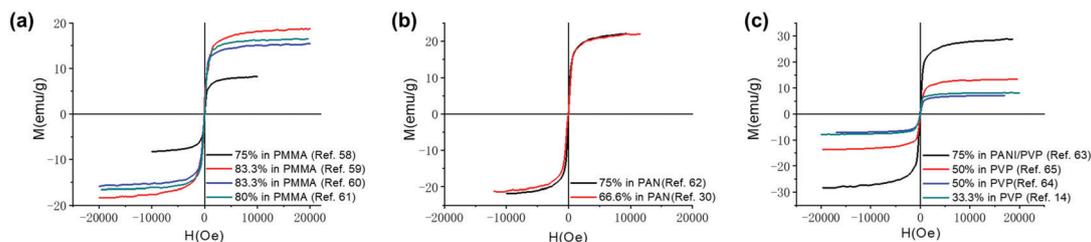
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Fig. 4 Magnetization curves produced when mixing Fe_3O_4 NPs with different polymeric materials (a) Fe_3O_4 NPs in PMMA,[63–66] (b) Fe_3O_4 NPs in PAN,[33,67] and (c) Fe_3O_4 NPs in PVP.[14,68–70]

and parallel-plate electrospinning. The method of electrospinning used affects the structure and properties of the composite material. The M_s of fibres prepared by uniaxial electrospinning increases with increasing mass of Fe_3O_4.[71,72] Savva et al.[73] prepared oleic acid-coated magnetite nanoparticles, which show lower saturation magnetisation (~ 40 emu g^{-1}) due to the presence of the organic, nonmagnetic oleic acid coating. However, no significant agglomeration phenomena occur during the electrospinning process as exhibited in Fig. 5(a).

Fluorescent magnetic NFs have been targeted in research owing to their suitability in a wide range of applications such as;

Fig. 5 (a) TEM bright field image of a PVP/PLLA/OA–Fe_3O_4 nanocomposite membrane;[73] (b) BM image of [Fe_3O_4/PMMA] coaxial nanobelts;[74] (c) BM image of [Fe_3O_4/PANI/PMMA]//[$Tb(BA)_3$phen/PMMA] Janus nanoribbons;[28] (d) FESEM image of PVP NFs;[44] (e) TEM image of α-Fe_2O_3/europium complex [Eu(DBM)$_3$(Bath), where DBM is dibenzoylmethanate and Bath is bathophenanthroline]/PVP composite NFs;[44] (f) TEM image of $CoFe_2O_4$/ yttrium aluminium garnet (YAG):5% Eu^{3+}/PVP composite NFs;[75] reproduced from ref. 75 with permission from The Royal Society of Chemistry; (g) TEM image of strontium hexaferrite nanoparticles (SrM-NPs) embedded in a PVA matrix;[76] (h) TEM image of SrM-NPs;[76] (i) TEM image of NiZn ferrite nanoparticles;[77] (j) SEM image of 1% MGNPs-polymer;[78] (k) SEM image of 3% MGNPs-polymer;[78] and (l) SEM images of 7% MGNPs-PEO.[78]

light-emitting diodes,[79] sensors,[80] resonators[81] and full-colour displays.[82] However, heavy losses in fluorescent intensity is observed when Fe_3O_4 NPs are in direct contact with luminescent compounds.[69] In order to circumnavigate this problem, core–shell and Janus structures have been produced as they offer the opportunity to incorporate both components in disparate zones of the material; minimising the direct interactions that would typically occur between them. Shao et al.[64] reported the fabrication of tuneable fluorescent colour-electrical-magnetic trifunctional coaxial nanoribbons using coaxial electrospinning. These coaxial nanoribbons exhibited similar magnetic properties (M_s of 18.58 emu g^{-1}) to the corresponding composite nanoribbons (where all components were mixed within the ribbons). Most significantly, the fluorescent intensity and electrical conductivity of the coaxial nanoribbons were considerably higher than those of the composite nanoribbons, demonstrating the importance of architecture derived properties. Fig. 5(b) demonstrates the coaxial nanobelt structure, revealing that the core contains large quantities of dark-coloured Fe_3O_4 NPs whilst the shell of the coaxial nanobelts appears transparent.[74]

Another effective method to create the Janus structure is via parallel-plate electrospinning. Gai et al.[83,84] prepared Janus nanobelts from Fe_3O_4/PVP and rare earth complex/PVP which demonstrated desired magnetism–luminescence bifunctionality. The M_s ranged from 3.16 emu g^{-1} to 10.19 emu g^{-1} and the results suggest that the magnetism can be tuned via different Fe_3O_4 NP loadings. The Janus nanobelts exhibited superparamagnetic behaviour using Fe_3O_4 nanoparticles of approximately 15 nm diameter. When the dimensions of the magnetic component, such as magnetite, drop to less than 20 nanometres, its magnetisation direction can flip randomly under the influence of temperature. However, in this circumstance, magnetite becomes superparamagnetic with only one magnetism domain.[85] In another example, Ma et al.[86] fabricated Janus NFs with Fe_3O_4/poly(methyl methacrylate) (PMMA) as the magnetic component and the M_s reached 32.61 emu g^{-1} when the mass ratio of Fe_3O_4 to PMMA was 6:1. This is similar to Fe_3O_4/rare earth complex/PMMA composite nanobelts (32.15 emu g^{-1}) that have also been produced.[86] However, the fluorescent intensity of the Janus nanobelts is considerably higher than that of the composite nanobelts. Additionally, luminescent–electrical–magnetic trifunctional materials are also a popular target structure in multifunctional nanocomposites. Lv et al.[87] added polyaniline (PANI) to the magnetic half of the Janus structure and the electrical conductivity

values of the Janus NFs increased with increasing PANI loading. However, the conductivity of the Janus NFs decreased with increasing amounts of Fe_3O_4 NPs due to the influence of Fe_3O_4 on the polymerisation process of aniline. The inner structure of the Janus nanoribbons can be revealed by the transmission light of a biological microscope (BM). As shown in the Fig. 5(c), one side of the Janus nanoribbon contains large quantities of dark coloured PANI and Fe_3O_4 NPs and, by contrast, the other side is transparent.[28]

2.2.2 α-Fe_2O_3 and other iron oxides. α-Fe_2O_3 is another popular choice of magnetic material due to its high stability, ease of fabrication, appropriate saturation magnetisation and increased acid resistance compared with Fe_3O_4. Additionally, α-Fe_2O_3 also has excellent adsorption ability of heavy metal ions and organic pollutants.[88] Wang et al.[44] fabricated NFs based on α-Fe_2O_3 nanoparticles and a europium complex to achieve magnetic–photoluminescent bifunctionality. The coercivity of the sample was 327 Oe with a remanent magnetisation (M_r/M_s) ratio of 0.244. Wang et al.[44] found that the $^5D_0$–$^7F_2$ transition was higher in the pure complex than in the composites. This was due to the addition of α-Fe_2O_3 nanoparticles decreasing the symmetry of the coordination environment for the Eu^{3+} ions. As shown in Fig. 5(d), the pure poly(vinyl alcohol) (PVA) NFs are relatively smooth. After incorporation of europium complexes and α-Fe_2O_3 nanoparticles into the polymer matrix the average diameter increased; this was due to an increase in viscosity of the feed solution.

Hematite is also often blended with polymeric materials via uniaxial electrospinning. Meng et al.[89] produced a paramagnetic nanofibrous composite film with polylactide (PLA), hydroxyapatite and γ-Fe_2O_3 nanoparticles. The M_s of γ-Fe_2O_3 NPs was 67.6 emu g^{-1} whilst the M_s of the film was 0.0492 emu g^{-1}, achieved at a mass ratio of 8.3% γ-Fe_2O_3 NPs within the film. Alternatively, Khanlou et al.[90] prepared γ-Fe_2O_3 NPs through a chemical co-precipitation process with an M_s of 12.19 emu g^{-1}. Following the chemical co-precipitation, at 5 wt% γ-Fe_2O_3 NPs, the NPs were added to a PMMA solution. The M_s of the composite produced was then 6.172 emu g^{-1}. Both cases demonstrate that the magnetic properties of MNP blended NFs are not proportional to the mass ratio of MNPs. Polymers do not simply act as a loading matrix but interact with MNPs and mutually influence the overall magnetic properties.

Additionally, there are other iron oxides that can be used to produce magnetic nanomaterials. For example Zhu et al.[91] produced core–shell Fe–FeO nanoparticles with an average diameter of 20 nm. The M_s of the Fe@FeO NPs was 108.1 emu g^{-1} whilst that of the nanocomposite fibres was 30.6 emu g^{-1}; with a nanoparticle loading of 30 wt%. Before electrospinning, the radii of the core and shell was calculated to be 13.2 and 6.8 nm, respectively. However, after electrospinning the radii became 12.7 (core) and 7.3 nm (shell). The shell thickness increase was attributed to an increase in particle oxidation at the extremely high voltages used during the electrospinning process.

Finally, Murillo-Ortíz et al.[76] embedded strontium hexaferrite nanoparticles (SrM-NPs) in PVA NFs. The ratio of M_r/M_s increased by 81 when 30 wt% SrM-NPs were added to the PVA solution.

As observed in Fig. 5(g), these nanoparticles have uniform size and have a localised distribution of NPs inside the surface of the NFs. Additionally, they do not show the presence of agglomerates. Fig. 5(h) then shows that the nanoparticles are ordered on the surface of the fibre and aligned with respect to the NFs' growth. This is a consequence of the nanoparticles' interaction with the highly intense electric field aligned with the electrodes in a point-plate configuration.

2.2.3 MFe_2O_4 (M = metal). In typical spinel and inverse spinel structures of magnetite, Fe(II) and Fe(III) ions are distributed in either octahedral or tetrahedral voids. MFe_2O_4 can be prepared by doping Mn^{2+}, Fe^{2+}, Co^{2+} and Ni^{2+} metal ions into the crystal voids of the iron oxide nanoparticles (IONPs).[92] This has resulted in nanoscale spinel ferrites MFe_2O_4 (M = Co, Mn, or Ni) with 1D structures (e.g. fibres) being studied more actively in recent years.[93]

Chen et al.[94] synthesised and modified $CoFe_2O_4$ nanoparticles to improve dispersion. The diameter of the $CoFe_2O_4$ particles produced was 5 nm and the M_s achieved was 50 emu g^{-1}. The diameter achieved is smaller than that of the bulk materials, due to the size of the $CoFe_2O_4$ crystallites and fewer defects being present in the structure. Finally, the $CoFe_2O_4$ NPs were mixed with polyacrylonitrile (PAN) and the composite exhibited an M_s of 45 emu g^{-1}. The decrease in M_s is attributed to the non-magnetic material coating (PAN) and its influence on the uniformity and magnitude of magnetisation by extinguishing the surface magnetic moment. Alternatively, Wang et al.[95] fabricated Janus NFs using $CoFe_2O_4$ to achieve magnetism–luminescence bifunctionality. When the mass ratio of $CoFe_2O_4$: PAN was 1 : 3 the M_s and H_c achieved were 5.09 memu g^{-1} and 20 kOe, respectively. Additionally, Bi et al.[75] electrospun $[Fe(NO_3)_3 + Co(NO_3)_2]$/PVP precursor solution before annealing in air at 700 °C for 4 hours to prepare $CoFe_2O_4$ NFs. YAG:5% Eu^{3+} calcinated NFs were also prepared via the same method. Janus NFs were then fabricated from both the $CoFe_2O_4$ NFs/PVP and YAG:5% Eu^{3+} NFs/PVP solutions, as shown in Fig. 5(f). The M_s of the $CoFe_2O_4$ NFs was 41.34 emu g^{-1} whilst the M_s of the Janus NFs ranged from 3.12–20.32 emu g^{-1}. The observed enhanced performance is attributed to the isolation of YAG:5% Eu^{3+} luminescent NFs from the $CoFe_2O_4$ magnetic NFs. Gonçalves et al.[96] prepared composite fibres of $CoFe_2O_4$ and poly(vinylidene fluoride) (PVDF). The composites demonstrated an increase in magnetisation with increasing $CoFe_2O_4$ content. They also found that the piezoelectric coefficient of the NF composites increased with increasing applied magnetic field. This is a result of the strain-mediated coupling between the magnetostrictive $CoFe_2O_4$ nanoparticles and the piezoelectric PVDF matrix. However, when compared with bulk polymers the piezoelectric coefficients were lower. It is speculated that this reduction is due to clamping by the surrounding material; which may significantly reduce the local deformation of the NFs.

Ghanbari et al.[97] synthesised $CaFe_2O_4$ nanoparticles that exhibit ferrimagnetism before producing cellulose acetate (CA)–Ag–$CaFe_2O_4$ nanocomposites by electrospinning. The M_s, of the nanoparticle compared to the NF, decreased from 6.1 to 0.31 emu g^{-1} whereas the H_c increased from 40 to 78 Oe, respectively. The authors stated

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that the magnetic moments of the $CaFe_2O_4$ nanoparticles are pinned by the polymer chains so that a higher magnetic field is required to align the single domain nanoparticles in the field direction. Additionally, Khan et al.[77] prepared $Ni_{0.6}Zn_{0.4}Fe_2O_4$ nanoparticles (see the transmission electron microscopy (TEM) image in Fig. 5(i)) with M_s of 26.81 emu g^{-1}. The NPs were then incorporated into composite NFs [with carbon nanotubes and recycled polystyrene (PS)] at 7.5, 15, and 30 wt% to produce fabrics with M_s values of 2, 4, and 8 emu g^{-1}, respectively.

2.2.4 Other magnetic materials. Other active magnetic materials that have also been incorporated into NPs include Ni, Co, and some dopants (such as Sr). Chen et al.[98] reported the M_s of pure Ni nanoparticles as 14.3 emu g^{-1}, which was considerably lower than bulk nickel (58.57 emu g^{-1}). The decrease is attributed to the oxidation of the nickel nanoparticles as a consequence of their large surface area. In another study, Gupta et al.[99] measured the M_s of pure MnZnFe–Ni as 25.47 emu g^{-1} before forming a blend with Estane® 5750; a polyester-based segmented polyurethane (PU). The M_s of the blend increased from 1.71 emu g^{-1} to 6.33 emu g^{-1}, with increasing MnZnFe–Ni content, and the composite NFs formed demonstrated superparamagnetic behaviour. Murillo-Ortíz et al.[17] synthesised $SrFe_{12}O_{19}$ with diameters ranging from 37 nm to 179 nm. The M_r/M_s and H_c of the $SrFe_{12}O_{19}$ were 0.63 and 6.22 kOe, respectively. The remanent squareness increased by 15% and the coercivity by 1.2% when $SrFe_{12}O_{19}$ nanoparticles were added to the PVA NFs by electrospinning. Additionally, the thermal stability and arsenic adsorption ability of the NFs was improved upon $SrFe_{12}O_{19}$ addition.

Erfan et al.[78] prepared ferrimagnetic glass ceramics, with a diameter of 10 nm, through the use of high-energy ball milling. The M_s of the magnetic glass ceramic nanoparticles (MGNPs) was 53 emu g^{-1} and the H_c equal to 88 Oe. The M_s of the composite fibre reached a maximum of 4.16 emu g^{-1} when the mass ratio of MGNPs was 7%. Low MGNPs concentration (1 wt%) NFs (Fig. 5(j)) appear clear and smooth, however, the roughness and nanoparticle aggregation on the surface of the NF increased at higher MGNP content (Fig. 5(k) (3 wt%) and 5 (l) (5 wt%)).

Min et al.[100] fabricated PVA/ferritin superparamagnetic fibres. The interaction between the host PVA hydrogel and the protein shell on the ferritin bio-nanoparticles was controlled by thermal methods to vary the size and concentration of the ferritin clusters. The average size and concentration of the ferritin clusters increased in the PVA NFs when the mixing temperature was raised from 30 to 80 °C. The close proximity of the ferritin cores within the clusters resulted in magnetic ordering and increased magnetisation in some cases.

Additionally, many research groups are now developing novel magnetic nanomaterials. One example is FePt which has demonstrated good chemical stability and high magnetocrystalline anisotropy.[101] Another example shows micron size graphene sheets decorated with cobalt NPs to endow magnetism.[102] There has also been an increasing focus on producing composite fibres with a range of polymeric materials. However, the mechanism regarding the interaction between the magnetic nanoparticles and polymer materials remains unclear.

3. Electrospun fibrous templates

This section discusses the various approaches used to create magnetic nanofibrous materials by electrospinning template constructs and the post-spinning treatment used: (i) deposition of magnetic NPs; (ii) carbonisation; and (iii) calcination. Fig. 1(b and c) highlights the different routes that have been used in this area of research. Table 2 serves as a comprehensive summary of this section, listing the various systems that have been explored, the magnetic reagent(s), polymer(s), electrospinning and processing approach, the final morphology obtained, major magnetic property and applications (where relevant) in each case.

3.1 Product morphology and corresponding processing

The template method is a cost-effective and scalable route used to produce both pure inorganic magnetic NFs and organic–inorganic hybrid magnetic NFs. Electrospun fibrous templating is also a convenient method for inorganic magnetic nanofibre (MNF) fabrication and the MNFs produced sometimes exhibit an increase in M_s.[103] However, the change in M_s depends on a range of complex factors and is not always enhanced. The appearance of a magnetically inert layer in the fibrous structure,[104] the formation of ferromagnetic phase[105] and smaller grain size are all factors that may contribute to a decrease in M_s in 1D nanostructures prepared by the electrospun fibrous template method.[106] The templating procedure of the MNFs (Fig. 1(b and c)) generally encompasses two steps: electrospinning to create a fibrous template followed by post-treatment of the template. A typical template-assisted electrospinning precursor solution is made up of three parts: (i) a magnetic component; (ii) polymer(s); and (iii) a solvent system. The magnetic component is typically in the form of magnetic nanoparticles (MNPs)[107] or metal salts. Polymers are added to the system to meet the viscosity demand of electrospinning (i.e. to provide entanglements). The most widely used polymers include PANI,[108] poly(ethylene oxide) (PEO),[109] polyurethane,[107] PVA,[110] PAN,[111,112] and PVP.[48,113–116] The purpose of the solvent is to dissolve the polymer(s), disperse or dissolve the magnetic component used and improve the charge-carrying capacity of solution. Typical solvents used are DMF, dichloromethane (DCM), methyl ethyl ketone (MEK), deionised water, ethanol[116] or a combination thereof.[107] The viscosity, concentration, surface tension and dielectric properties of the spinning solution exert a great influence on the electrospinnability of the solution and the diameter, morphology, crystallinity and tensile strength of the NFs. Such influences have been referred to in previous studies.[117,118] Finally, Fig. 1(c) highlights an alternative route where NFs without any magnetic components are used as templates before MNPs are deposited upon them.

In the sol–gel process, literature reports have demonstrated the use of metal nitrate[114,116,119] and acetate[114] precursors with polymer(s) in the production of electrospun NFs with uniform distributions, even at very high loadings.[120] Although the mechanisms for such processes were not explicitly discussed, the conditions in which the precursor solutions were prepared are reported.[48,121,122]

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Following initial deposition, the obtained fibres are often dried and pre-sintered to evaporate any solvents.[94,114–116,119,121,123] This is followed by calcination, at a given heating rate, in order to (partially or completely) remove the polymers to form pure inorganic fibres. Concomitantly, the metal ions are converted to their neutral elemental state[48] and the atoms aggregate to form nanoparticles.[124] The calcination conditions are adapted to meet the varying demands of a given system. For example, air is an idealised atmosphere if metal oxide NFs[125] are targeted, whereas hydrogen (H_2),[126–128] often in combination with an inert gas such as argon (Ar),[129–132] and ammonia gas (NH_3) are adopted as a reduction (often referred to as de-oxidation in the literature) source. In some cases, the calcination atmosphere is selected to generate oxygen vacancies or eliminate impure phases.[133] For example, α-Fe_2O_3 NFs obtained in a study by Guo *et al.*[134] were heated in an NH_3 atmosphere and the crystallites were transformed into Fe_3O_4 crystallites. Additionally, when the temperature was raised, Fe_2N NFs were obtained. The electrospinning parameters, environmental conditions, physical and electrical properties of the spinning solution, and calcinating conditions (temperature, rate, profile and duration) all enable the crystal structure, morphology and ultimate properties of the final product to be manipulated.[135] For example, the preparation of mullite–nickel nanocomposite NFs[132] was performed through two heating stages. The first was used to convert Ni^{2+} to Ni NPs in a reducing atmosphere between 550–750 °C before the mullite phase was formed at 1000 °C. However, if the first stage was not allowed to proceed for sufficient time (to permit Ni^{2+} to be fully reduced to Ni NPs), an undesirable spinel phase mixed with a mullite phase was formed during the second heating stage. In a similar procedure, a Fe/Ni alloy was prepared from the reduction of $NiFe_2O_4$ (300–600 °C) to produce Fe–Ni alloy nanoribbons.[126,136] The morphology of the Fe–Ni alloy precursor (body-centred cubic or face-centred cubic) was influenced by the heating temperature used during the reduction stage.

Often the calcinated products are formed as powders and do not maintain their structure.[41] Efforts have been made to resolve this problem by shifting towards the production of fibre mats that offer the added benefit of being flexible, easy to produce, recyclable and cost-effective. In one example, flexible fibre mats have been utilised in the development of magnetic devices; offering a new design method for electromagnetic shielding systems.[121] Additionally, the ease of recycling the mats and the enhanced structural stability, when compared to powders, is highly desirable in applications such as waste water treatment.[118] The key factors that influence the flexibility of these fibrous mats are the uniform distribution of NPs throughout the NFs,[110] the porous structure of the mats[110] and the interfacial energy between NPs and NFs.[110] Consequently, attempts have been made to improve the interfacial interactions between the MNPs and the matrix.[94] For example, zein (a maize protein) was used as an adhesive between $NiFe_2O_4$ NPs and SiO_2 NFs to render the resultant fibre mats with extraordinary flexibility.[110] In addition, Wang *et al.* reported that oxygen plasma treatment was an effective way to keep the membrane integrated during the stage where the organic component is removed.[41] Finally, flexible calcinated γ-Fe_2O_3/C NF mats have
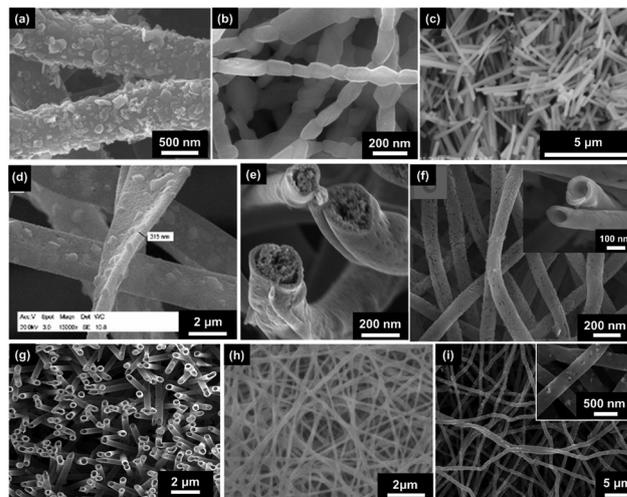
Fig. 6 Example SEM images of: (a) fibres with particle deposition;[107] (b) necklace-like NFs;[138] (c) short NFs;[139] (d) nanobelts;[111] (e) core–shell NFs;[140] reproduced from ref. 140 with permission from The Royal Society of Chemistry; (f) hollow NFs;[134] reproduced from ref. 134 with permission from The Royal Society of Chemistry; (g) nanotubes;[141] reprinted with permission from ref. 141. Copyright (2021) American Chemical Society; (h) 3D crosslinking NFs;[109] and (i) regular NFs (calcinated).[142] Specific details: (a) Fe_3O_4@PU NFs, Fe_3O_4 NPs at 1 mg ml^{-1}; (b) $SrFe_{12}O_{19}$ NFs calcinated at 1000 °C; (c) $BiFeO_3$ NFs calcinated at 550 °C; (d) Fe_3O_4/C NFs calcinated at 800 °C; (e) $CoFe_2O_4$–$Pb(Zr_{0.52}Ti_{0.48})O_3$ NFs calcinated at 750 °C; (f) Fe_2N NFs calcinated at 400 °C; (g) SnO_2 NFs calcinated at 500 °C; (h) Fe_3O_4–alginate (SA)/PVA crosslinked NFs; and (i) $NiFe_2O_4$/multi-walled carbon nanotube (MWCNT) carbon-based NFs (CNFs) calcinated at 850 °C.

also been prepared by combining electrospinning, hydrothermal synthesis and calcination.[121,137]

NFs with different morphologies can also be fabricated by changing the shape of the liquid supply nozzle or by adjusting the calcination parameters used to remove the polymer construct. In the first stage, when polymers are present, fibres decorated with inorganic particles are obtained. Once the polymers have been removed, necklace like NFs, short NFs, nanobelts, core shell NFs, hollow NFs and regular NFs can be obtained. Fig. 6 shows the various types of MNFs that can be targeted by electrospinning, as discussed in the following sections.

3.1.1 Decorated fibres. Fig. 1(c) shows how fibres decorated with particles can be created and Fig. 6(a) shows an example of such surface topology. Polymers are electrospun to form fibres and then MNPs are coated onto the fibres either by *in situ* synthesis[109,143] or post-spinning deposition.[107,108,112] Both approaches face the issue of aggregation, however, there have been few research efforts focused on resolving this problem.[107,110,143] In one notable example, *in situ* synthesis was studied, focussing on the influence of drying mode on the morphology of iron oxides upon polyimide (PI) NFs.[143] This study concluded that drying under vacuum was more effective than drying in air to allow for uniform particle distribution. When using the deposition method, specific techniques are used to provide uniform distribution of the MNPs. These techniques include use of a surfactant,[110] polyol immersion[107] and layer-by-layer assembly.[144] In some cases, where the polymer completely remains, cross-linking agents

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can be used to form 3D cross-linked networks[109] to improve water resistance and mechanical integrity.[145]

3.1.2 Necklace-like nanofibres. In single nozzle electro-spinning, where amorphous polymer templates are removed by calcination, NFs will inevitably transform from smooth morphologies to rough surfaces before necklace-like structures are eventually formed (Fig. 6(b)). This happens when the calcination temperature reaches a relatively high value. For example, Xiang et al.[122] found that the calcinated $Co_{0.5}Ni_{0.5}Fe_2O_4$ NFs formed the necklace-like morphology when the temperature was greater than 800 °C. The transmission electron microscopy (TEM) images collected in this study revealed that the rapid growth of nanocrystals, aggregation of small NPs and resultant formation of larger NPs at high temperatures gave rise to their unique morphology.

3.1.3 Short nanofibres. The length of electrospun NFs can reach several tens to even hundreds of microns,[146,147] where 'short' NFs are defined as those that are less than ten microns in length (Fig. 6(c)). Short/broken fibres are typically generated under two circumstances: (1) when the NPs experience uneven thermal stress and they aggregate; and (2) by the inhomogeneous distribution of sacrificial templates.[139,148] In both cases, the local area will fracture readily. For example, Sakar et al.[139] fabricated $BiFeO_3$ NFs at varied spinning voltages (8, 10, 15 and 20 kV) and found that 10 kV or less was not sufficient to homogenously distribute the polymer. The calcinated products then have the tendency to be broken at the area where the polymer is scarce.

3.1.4 Nanobelts and nanoribbons. Necklace-like NFs (Fig. 6(b)), nanobelts (Fig. 6(d)) or nanoribbons can be fabricated with minor adjustment to the composition of the spinning solution and electrospinning experimental conditions.[111] One possible formation mechanism suggests that if the solvent evaporates rapidly during jet flow, a columnar transition state structure is formed with a thin elastic polymer skin. This structure buckles gradually and finally becomes belt-like.[119,149] Additionally, in this process, specific polymers (such as PMMA) are found to be beneficial towards the formation of the flat shape belt NF morphology.[111] Additionally, the effects of calcination have been investigated and show no significant changes to the morphology of the nanobelt following removal of the polymer.

3.1.5 Core–shell nanofibres. An example of the core–shell structure is illustrated in Fig. 6(e). These structures are typically generated by coaxial electrospinning or, alternatively, by using a single nozzle, followed by calcination and finally coating. There have been a number of reports where magnetic core–shell NFs are produced. In one example, Bayat et al. coated polydimethylsiloxane (PDMS) on Fe_3O_4/C NFs, which were produced via electrospinning followed by pyrolysis at 900 °C, for future use as electromagnetic interference (EMI) shields.[150] In a second method, α-Fe_2O_3 NFs were soaked in a tetraethyl orthosilicate (TEOS) precursor solution whereby a catalysed hydrolysis and condensation polymerisation reaction resulted in a uniform silica (SiO_2) coating along the fibres.[151] Li et al.[3] and Zou et al.[152] coated $SrFe_{12}O_{19}$ and $CuFe_2O_4$ NFs with TiO_2 and CeO_2, respectively, via a soaking and calcination procedure. In addition to these multi-step procedures, as aforementioned,

core–shell NFs can be fabricated through a straightforward one-step method by coaxial electrospinning. In this case, two different spinning solutions are loaded into inner and outer syringes. Each jet flow is then formed with a different velocity and collected before being annealed to obtain magnetic–magnetic,[153–155] magnetic–luminescent[156] or magnetic–electric[48,140,157] bifunctional core–shell nanofibrous materials.

3.1.6 Hollow nanofibres. Similar to the construction of core–shell MNFs, hollow fibres (Fig. 6(f)) can be prepared with single[158–160] or coaxial nozzles,[15,161] both followed by calcination. During the single spinning procedure, both Cheng et al.[160] and Chen et al.[98] identify the key factor critical in hollow or solid morphology determination. Cheng et al. shows that the gas diffusion rate, from the interspaces between the nanoparticles, is slower than the PVP decomposition. As the polymer decomposition continues, the observed increase in internal pressure forces the nanoparticles to the exterior and hollow fibres form.[160] However, the study of Chen et al. highlighted that metallic shell formation is the key factor in hollow morphology determination.[103,158] When the NFs are heated in air, metal oxide clusters begin to form on the surface of the fibre until they create a shell. After which, at even higher temperatures, the polymer component degrades to form volatile decomposition products leaving behind the hollow NF structure. This illustrates that hollow structures are difficult to produce in cases where the metal oxide forming temperature is higher than the decomposition temperature of the polymer matrix. Further studies by Zhao et al.[159] have determined that a concentration gradient (which causes varied consumption rates of Cu^{2+} and Fe^{3+}) and difference in gas pressure between the inside and outside of the fibre are also two important factors to consider when preparing hollow $CuFe_2O_4$ NFs. On the other hand, Fu et al. determined that phase separation caused by a concentration gradient is the most critical factor in the preparation of hollow MNFs.[135] Notably, the organic nanofibrous template can be removed afterwards to create hollow MNFs.[162]

Nanotubes (Fig. 6(g)) are similar in structure to hollow fibres and the differences between them are subtle and sometimes difficult to identify.[163] In some cases, the term 'nanotube' is used to describe shorter, defect-free hollow fibres, but there is no clear definition and the term tends to be used interchangeably with hollow NFs across the literature. Accurate heating methods are required for nanotube construction via single nozzle electrospinning and Jiang et al. has reported the successful synthesis of nanotubes several microns in length.[164] Additionally, Li et al. obtained $NaYF_4$:Yb/Er/Gd-decorated SiO_2 nanotubes via single nozzle electrospinning followed by calcination using a spinning solution comprised of $NaYF_4$:Yb/Er/Gd nanocrystals, TEOS and PVP. The as-spun NFs were annealed between 200–600 °C at a heating rate of 2 °C min⁻¹. In this example, PVP is forced to migrate to the surface once leaving the nozzle resulting in a PVP–silica shell with the magnetic component and TEOS within the core. Upon removal of the organic material, nanotubes were produced.[165]

In the coaxial spinning procedure, the pure organic part (such as mineral oil, polymers or components that incorporate magnetic ingredients[166]) can be adopted as the core solution.[167]

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However, the flow rate of the inner solution should be controlled to prevent leakage.[166]

In addition to the aforementioned traditional single-walled hollow structures, there are novel structures that have been obtained, such as hollow-core–double-shell NFs. Kim *et al.*[158] produced $CoFe_2O_4$ core–PANI double shell structures. Initially hollow $CoFe_2O_4$ NFs were obtained by calcinating the precursor fibres between 80–550 °C at a heating rate of 5 °C min^{-1} for 2 hours. The inner and outer surfaces were then coated with PANI *via in situ* oxidative polymerisation.

3.1.7 Three dimensional networks. Post-treatment can be adopted to enhance the properties of NF mats. If the polymer scaffold is retained in the final product, a cross-linking agent is frequently used to form 3D cross-linked networks[109] (Fig. 6(h)) to improve water resistance and enhance mechanical stability.[145] For example, Gao *et al.* adopted N,N'-trimethylene-bis[2-(vinylsulfonyl)-acetamide] as a chemical cross-linking agent for PVA. Whilst a slight weight loss ($<0.4\%$) was observed after immersing the mat in hot water for two hours the cross-linked PVA NF mats demonstrated enhanced water resistance.[168] Chemically cross-linking the polymer chains prevents dissolution of the macromolecules and changes the mechanical properties of the constructs, due to the creation of a chemically bound mesh that no longer relies on physical entanglements for structural integrity.

3.1.8 Regular nanofibres. If the NFs are calcinated solid fibres that cannot be categorised into any of the aforementioned morphologies, they are defined as regular NFs (Fig. 6(i)). Regular NFs are then subdivided into two categories, depending on whether they have rough or smooth surface morphology. Generally, aggregated crystals consisting of NFs appear after the removal of organic components and result in the surface morphology of the NFs being rough. However, there are two exceptions to this rule. The first case is when carbon-containing backbone polymers are carbonised instead of being completely removed. In this case, MNPs are embedded in the continuous carbon phase.[94] The second case is when the calcination atmosphere is controlled. For example, Barakat *et al.*[169] revealed that smooth morphologies of nickel NFs were retained even after heat treatment in argon. However, previous studies where hydrogen has been adopted to reduce the nickel precursor NFs do not produce a smooth morphology. To obtain NFs with a uniform size distribution, the surface tension, viscosity and electronic conductivity can be adjusted through the addition of a low molecular weight agent, if optimum conditions cannot be otherwise identified.[168]

3.2 Processing parameters and magnetic properties

The most common processing parameter for electrospinning NF templates is calcination. Calcinated NFs still possess superior magnetic performance than their zero-dimensional (0D, *i.e.* nanoparticles) or bulk counterparts. The most apparent property enhancement is in the coercivity (H_c) due to the anisotropic nature of NFs, as compared to NPs.[127] The magnitude of the coercivity is related to the anisotropy of the material, which includes magnetocrystalline anisotropy, shape anisotropy and stress anisotropy. However, the NF structure possesses large shape anisotropy as a
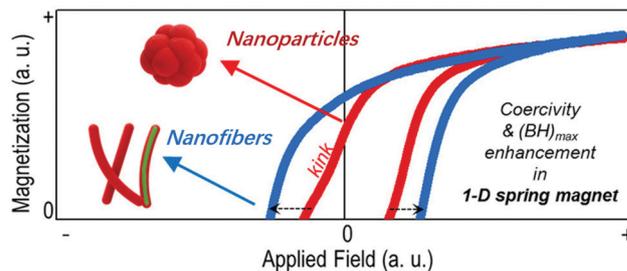

Fig. 7 Scheme of expected magnetic performance in a uniform, one-dimensional, hard–soft magnetic core–shell nanocomposite and its counterpart.[154] Adapted with permission from ref. 154. Copyright (2021) American Chemical Society.

result of the strict restriction of magnetic moment along the fibre axis.[170] For example, $LaFeO_3$ exhibits antiferromagnetism in the bulk, yet is ferromagnetic when in processed as a one-dimensional (1D) nanostructure with an increased coercivity value of 28 000 Oe.[171] In addition, 1D hard–soft exchange-coupling nanomaterials offer a unique platform for enhancing the maximum magnetic energy product [(BH)max] due to the reduced self-aggregation when compared to their 0D counterparts.[124,154] In another case, Lee *et al.*[154] fabricated core–shell Sm_2Co_{17}/FeCo NFs by electrospinning, calcination, calciothermic reduction and electroless plating. Relative to its 0D counterpart, the well-dispersed 1D/core–shell nanostructure demonstrates an enhancement in (BH)max (46% increase) as a result of the dense homogeneous soft magnetic coating as shown in Fig. 7. However, it should be noted that the magnetic properties are not always improved by simply introducing anisotropy (going from nanoparticles (0D) to NFs (1D) nanomaterials). This was highlighted in a study where the M_s value of $CoFe_2O_4$ NFs was lower than that of $CoFe_2O_4$ NPs and even the bulk counterparts.[172]

3.2.1 Morphology. As aforementioned, morphology is another factor that influences the magnetic properties of the final NFs. Manipulating the electrospinning voltage can cause the morphology of the MNFs to go through a series of changes. However, this has only been demonstrated in a limited number of studies. In one example, at lower voltages (~ 8 kV) the NFs aggregate and by gradually increasing the voltage pass through the following formations: rods (~ 10 kV), fibres (~ 15 kV) and finally, belts (~ 20 kV). The M_s values are then shown to decrease in the sequence of belts $>$ rods $>$ fibres $>$ aggregated fibres.[139] In another notable example, Zhao *et al.* highlighted that SnO_2 nanotubes exhibit stronger room-temperature ferromagnetism than the corresponding NFs. This observation is attributed to a larger number of surface defects that result from the Sn interstitial and O vacancies having lower formation energies than the Sn vacancy and the surface area-to-volume atom quantities for the nanotubes being greater.[173] Additionally, the average particle size for the nanotubes (30 nm) was larger than that of the NFs (20 nm). Similarly, the average outer diameter of the nanotubes (110 nm, with a 10 nm wall thickness) was also larger than that of the NFs (100 nm).

3.2.2 Calcination. The magnetic properties of NFs are strongly dependent on calcination. Diamagnetic as-spun NFs

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have been shown to exhibit ferromagnetic behaviour after calcination. The calcination parameters such as rate, temperature and calcination atmosphere exert influence on the final properties of the NFs.[174] When the organic components are completely removed, the calcinated products are typically composed of nano-crystallites of various size. Overall, the magnetic behaviour of the nanocrystallites assembly is determined by the shape and size of the crystalline NPs and the interactions between them.[114,120,159,175]

3.2.2.1 Calcination temperature. The relationship between calcination temperature and M_s is often system dependent. Generally, for a given system, lower thermal treatment temperatures result in smaller particles being formed (down to a critical minimum size) and the smaller particles exhibit smaller M_s values.[106,114,119,122,132,133,170,172,176–188] Larger particles have less magnetic contribution from the surface (whilst the contribution from the interior concomitantly increases) and thus the M_s value increases.[114] However, the smaller M_s values observed at low calcination temperatures are not always only due to the decrease in size of the particles. For example, in Liu's study, the $CoFe_2O_4$ NFs were calcinated at 700, 750, 800, 850 °C, respectively. In addition to the increase in particle size, the XPS results suggested that as the calcination temperature increases, the migration of cobalt ions to the tetrahedral sites as well as iron ions to the octahedral sites increases, thus also contributing to the larger M_s observed.[189] However, there are examples that do deviate from this trend. For example, the M_s values of $CuCo_2O_4$ fibres have been found to decrease with increasing calcination temperature.[190] In this case, the decrease is attributed to surface distortions brought about by interactions between oxygen atoms and transition metal ions. $Ti_{0.9}V_{0.1}O_2$ NFs were also shown to exhibit a decrease in M_s with increasing heat treatment temperature, which was assigned to the increase of nearest-neighbour V ions with direct antiferromagnetic coupling.[191]

In the study of Liu *et al.*, the M_s value of α-Fe_2O_3/TiO_2 NFs decreased with increasing calcination temperature. The authors ascribed this to an increased density of α-Fe_2O_3.[177] The decrease in M_s with arising calcination temperature was also found in other studies.[192] In more recent work, Ponhan *et al.*[193] determined that the M_s dependency on calcination temperature of $ZnFe_2O_4$ NFs exhibits through both of the observed trends. Initially, M_s increases with calcination temperature before then decreasing, with crystallite size increasing typically from 19 to 26 nm.

Similarly, the H_c transition with calcination temperature is also system dependent. In most cases, H_c increases with calcination temperature; including that observed for $MgFe_2O_4$ nanotubes,[135] $CoFe_2O_4$ NFs,[194] $NiFe_2O_4$ NFs,[183] $MnFe_2O_4$ NFs,[186] Ni/mullite NFs[132] and $Co_{0.5}Ni_{0.5}Fe_2O_4$ NFs.[122] However, there are also reports where the H_c transition does not follow this trend. For example, the H_c value of $MnFe_2O_4$/C NFs decreases with increasing calcination temperature[185] and is attributed to the $MnFe_2O_4$/C NFs exhibiting ferromagnetic properties due to the distribution of cations over tetrahedral and octahedral sites. In the study by Lu *et al.*,[106] the H_c value of $CoFe_2O_4$ NFs demonstrated an initial increase followed by a decrease with increasing annealing temperature. The same phenomenon was

reported in both $SrTiO_3$/$SrFe_{12}O_{19}$ NFs[181] and yttrium iron garnet NFs,[170] attributed to the motion of domains experienced at different particle sizes. Furthermore, the same trend was found in hollow $CuFe_2O_4$ NFs.[159] In this case, higher calcination temperatures gave rise to larger MNPs and thus a higher field force was required to alter the magnetic moment direction; in turn resulting in higher H_c. However, when the particle size exceeded the critical size for a single-multi domain transition, H_c declined. Findings from the study of $CoFe_2O_4$ hollow fibres further support this claim as H_c increases with annealing temperature.[160] The H_c value of the $CoFe_2O_4$ nanobelts[119] achieved a maximum value of 1802 Oe before decreasing with further increases in calcination temperature as a result of the particle size being below the critical domain size. In addition, the absence of domain walls and unique rectangular cross-sectional shape contributed to the high magneto-crystalline properties observed.

The final important magnetic property to be discussed in relation to calcination temperature is remanent magnetisation, M_r. In one example, $CoFe_2O_4$ NFs, fabricated by single spinning and calcination, and were shown to exhibit a uniform increase in M_r with increasing calcination temperature. This is commonly attributed to an increase in the particle size.[194] The same phenomenon was found for $MnFe_2O_4$ NFs,[186] $SrTiO_3$/$SrFe_{12}O_{19}$ NFs,[181] $MgFe_2O_4$ nanotubes,[135] hollow $CuFe_2O_4$ NFs[159] and hollow $CoFe_2O_4$ NFs.[160] In one contradictory case, the M_r value of $CuCo_2O_4$ fibres[190] was found to decrease with increasing calcination temperature, which was suggested to be due to weakening of the superexchange interaction between the Cu^{3+} and Co^{2+} cations.

In some fabrication methodologies, the heat treatment encompasses two stages; calcination and reduction. The magnetic properties are known to be dependent on the reduction temperature. For example, the reduction temperature of Fe/Ni alloy nanobelts[126] show an increase followed by decreasing trend in M_s, whereas the H_c initially decreases before remaining almost constant with increasing temperature. In this example, the Fe/Ni alloy nanobelts were reduced from $NiFe_2O_4$ and the initial relative low temperatures used resulted in the observed partial reduction of M_s.

The relationship between calcination temperature and M_s, H_c and M_r has been identified as significantly system dependent. Factors such as density, surface distortions, cation distribution, particle size and cation interaction have all been shown to contribute to the final magnetic properties of the NFs produced.

3.2.2.2 Calcination atmosphere. The calcination atmosphere used can have a significant effect on the magnetic properties of NFs. For example, $SrFe_{12}O_{19}$/$CoFe_2O_4$ obtained from as-spun NFs calcinated in air showed a H_c value of 3190 Oe. On the other hand, $SrFe_{12}O_{19}$/FeCo obtained from as-spun NFs calcinated in air and then reduced in hydrogen showed a significantly lower H_c value of 1249 Oe. This change is attributed to the close contact of the soft and hard phases which in turn results in strong exchange coupling. In addition, Fe_2O_3 hollow NFs obtained from as-spun NFs calcinated in air were superpara-magnetic and achieved an M_s value of 4.34 emu g^{-1}, whilst

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Fe_3O_4 and Fe_2N hollow NFs obtained from as-spun NFs calcinated in air and then reduced in gaseous ammonia achieved M_s values of 82.99 emu g^{-1} and 2.07 emu g^{-1}, respectively. Finally, the Fe_3O_4 hollow NFs exhibited ferromagnetism whilst the Fe_2N hollow NFs demonstrated superparamagnetism.[134,153] The morphology of the fibre can also be altered depending on the calcination atmosphere used and, as previously discussed, is directly related to the magnetic properties of the final fibre. In one example α-Fe_2O_3 nanotubes and Fe_3O_4 NFs were produced when the calcination atmospheres were air and argon, respectively. Nanotubes were formed due to the lack of oxygen present as gas diffusion from the decomposed organic component (PEO) drives the nanoparticles from the inside to the outside of the fibres.[195]

3.2.3 Magnetic component loading. The most influential factor on the overall magnetic performance of NFs are the components used to make them, rather than how they are made. This is particularly evident in cases that involve the combination of magnetic inorganic constituents and non-magnetic polymers. Typically, M_s has a linear correlation with the quantity of magnetic components,[107,156,196–199] whilst the addition of non-magnetic components generally impairs the M_s value.[200–202] Therefore, the M_s values of NFs where polymers are retained are lower than those observed for fibres that consist of pure MNPs.[94,107] Another common example, reflecting this logical observation, is M_s enhancement after carbonisation. Again, due to the loss of the non-magnetic compounds during the procedure. With ascending carbonisation temperature, M_s will increase when the size and coarsening of MNPs within the NFs increases and if a new phase with larger M_s comes into force.[198] However, one report claims that the M_s value in their system was independent of shape and size of the MNPs and is only affected by the total quantity of magnetic atoms.[198]

Moreover, carbonisation often induces the transformation from superparamagnetism to ferromagnetism in electrospun NFs.[194] This is due to the formation of a new phase, such as the carbide crystal, which destroys the single crystal domain of the original NFs.[172] It has also been claimed that enhancement of particle size leads to higher magnetocrystalline anisotropy.[197,198]

3.2.4 Types of polymer. The different types of sacrificial polymer result in varied magnetic properties in the final NF. For example, NiO NFs obtained from poly(2-ethyl-2-oxazoline) (PEtO$_x$)/nickel(II) acetate tetrahydrate precursor were shown to exhibit antiferromagnetism, whereas NiO NFs obtained from styrene–acrylonitrile random copolymer (SAN)/nickel(II) acetate tetrahydrate precursor exhibited ferromagnetism at RT.[203] The authors suggested that the change in magnetism occurs because of the distinct degradation mechanism of the different polymers. The degradation of SAN begins with elimination of the nitrile group followed by random scission of the backbone. In contrast, PEtO$_x$ begins with random chain scission followed by elimination of the side chains. Additionally, the activation energy for grain growth and corresponding grain sizes were also different. Both factors lead to non-stoichiometry of the NiO NFs.

3.2.5 Polymer concentration. Polymer concentration notably influences the electrospinning process, which determines the morphology and magnetic properties of the fibres produced.

Typically, increasing the concentration of the polymer produces fibres with increased diameter, uniformity and tensile strength.[204] The variation in morphology, particularly NF diameter and uniformity, is also often attributed to the solution properties (polymer and solvent) on the electrospinning process; most significantly the viscosity of the electrospinning solution. The viscosity, alongside the surface tension, ultimately determine the ability of a given solution to be electrospun. Therefore, it is unsurprising to see examples in the literature that describe increased solution viscosity due to higher concentrations of polymer resulting in NFs with increased diameter.[205] In one investigation, the effect of PVP concentration on the morphology and magnetic properties of $SrFe_{12}O_{19}$ nanobelts was investigated.[116] It was found that increasing the PVP concentration from 8.5–12.3 wt% led to the increase in the width of the nanobelts produced. Additionally, the diameter of NPs were reduced and the NPs' overall size was largely below the single domain size. As a direct result of this, H_c achieved a maximum value of 7310 Oe (the highest known value in the literature to the best of our knowledge) of all the pure 1D $SrFe_{12}O_{19}$ structures. This value is also close to the theoretical limit.[116]

3.2.6 Dopants. Doping has been shown to affect the crystal size of magnetic materials, which influences the magnetic properties of the final MNFs. The extent of this effect depends on the properties and loading of the dopant.[206–212] Typical dopants that have been explored to enhance the magnetic properties and direct the morphology of MNFs include graphene oxide (GO),[213] Co,[214] Ni,[215] Cu[210] and Zn.[216] For example, GO-doped $Ni_{0.8}Gd_{0.2}Fe_2O_4$ NFs showed a transformation from fibre to ribbon morphology and a decrease in M_s with increasing GO content. Additionally, Ni-doped Fe NFs were shown to exhibit lower M_s and H_c values compared to pure Fe NFs.[128] Ni-doped $CoFe_2O_4$ NFs were also found to show a decreasing trend in M_s and H_c with increasing Ni^{2+} concentration, which was due to the lower magnetic moment and magnetocrystalline anisotropy of Ni^{2+} compared to Co^{2+} ions.[215] It is important to note that the change in magnetic properties with dopant concentration is not always uniform[209] and can contradict known theories (such as Brown's theory[217]) where M_s and H_c are inversely related. For example, the M_s value of Zn-doped $CuFe_2O_4$ NFs initially increases before decreasing with increasing Zn^{2+} concentration due to increased exchange interaction between ions within the lattice.[216] Co-Doped ZnO NFs were fabricated by electrospinning a PVA/Co-doped ZnO suspension followed by sintering in air.[218] Doping ZnO NFs with 1.8–7.2 wt% Co was shown to change the material properties from ferromagnetism to ferrimagnetism, as shown by the temperature-dependent magnetisation curves and Curie–Weiss fits in Fig. 8. The origin of ferrimagnetism in $Zn_{1-x}Co_xO$ NFs is attributed to the substitution of Co for Zn and the corresponding change of characteristics in the electronic state.

Doping has also been known to lead to the production of NFs with unique properties. For example, hole-doped manganites, with the general formula $R_{1-x}A_xMnO_3$ (where R refers to a trivalent rare earth element and A is a divalent alkaline earth element) were found to exhibit colossal magneto resistance (CMR). $La_{0.33}Pr_{0.34}Ca_{0.33}MnO_3$ NFs (prepared at a calcination temperature of 600 °C) exhibited
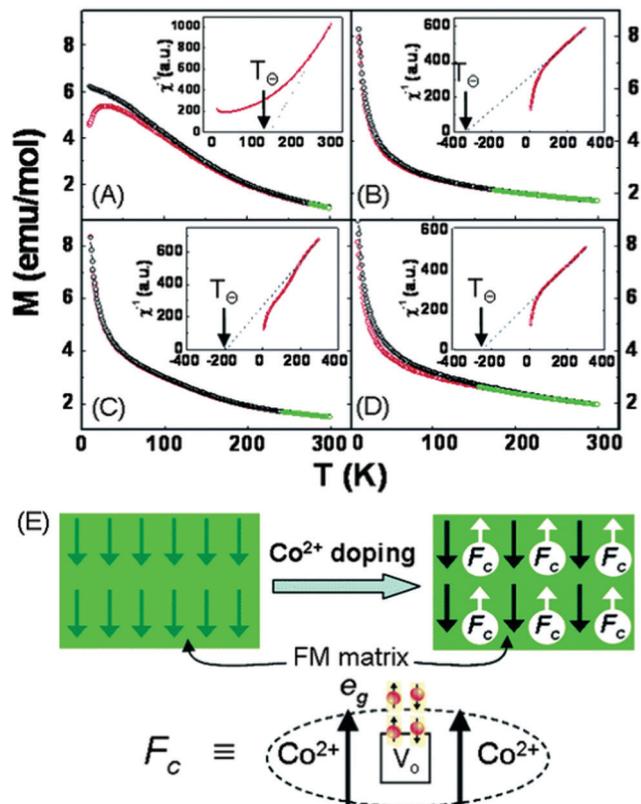
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Fig. 8 The effect of doping ZnO NFs with Co^{2+} on magnetic properties, illustrated through: (A)–(D) temperature-dependent magnetisation (M–T) curves of $Zn_{1-x}Co_xO$ ($x = 0$, pure ZnO nanowires) NFs with 1.8, 4.4 and 7.2% Co, respectively. This visually shows the transition from ferromagnetism to ferrimagnetism. (E) Co^{2+} doping-induced ferromagnetism to ferrimagnetism crossover.[218] Reproduced from ref. 218 with permission from The Royal Society of Chemistry.

CMR over a wide temperature range and the magneto resistance reached a maximum value of 95% at the metal–insulator transition temperature of 70 K under 7 Tesla.[219] In another case, doping α-Fe_2O_3 nanotubes with V_2O_5 was found to increase M_s, and decrease H_c, which was attributed to the aforementioned Brown's theory; where H_c and M_s are inversely proportional to one another.[217] In other work, after Co doping, $SrTiO_3$ was endowed with room temperature ferromagnetism due to Co addition and oxygen vacancies,[130,173] which has also been demonstrated in the study of SnO_2 nanotubes.[220] However, Mohanapriya et al. suggests that ferromagnetism in Mn-doped SnO_2 NFs is induced only by precipitated impurity phases instead of pure SnO_2 or dopant.[221] Nevertheless, Mn doping was found to impair the ferromagnetism of $BaTiO_3$ and resulted in dia- and paramagnetic behaviour of the $BaTi_{0.9}Mn_{0.1}O_3$ NFs that were produced.[222] Fe-Doped NiO NFs were prepared as a diluted magnetic semiconductor (DMS) and, notably, doping was shown not to affect the fibre diameter or surface morphology.[223] Finally, the addition of GO dopant to $CoFe_2O_4$ NFs demonstrated enhanced crystallinity, which boosted M_s values at low loadings, but was then shown to exhibit a slight decrease in M_s with higher GO loadings.[224]

3.2.7 Alloys. Alloy systems with nanoscale dimensions often demonstrate outstanding magnetic properties. There have been a number of reports where alloys have been processed into NFs to create magnetic materials. For example, Zhang et al.[128] reported Fe–Ni NFs for the first time and demonstrated that the use of an Fe–Ni alloy resulted in superior ferromagnetic properties. Additionally, it was suggested by Jiang et al.[126] that an Fe–Ni alloy also produces NFs with high saturation magnetisation, permeability, Curie temperature (T_c), low coercivity and low energy loss. FeCo is another promising soft magnetic nanomaterial which exhibits low magneto-striction, high saturation magnetisation, high resistivity, small coercive forces, high Curie temperature and high magnetic anisotropy energies.[225] Finally, a range of novel Sm_2Co_{17} NFs were fabricated by Lee et al. reporting a decrease in M_s and concomitant H_c increase with arising Sm content within the alloy.[206]

4. Applications

In this section, the application of magnetic materials, and how this relates to their final properties, are explored. This aptly highlights the versatility of electrospun magnetic materials whilst concomitantly highlighting the need for future work into their production and design.

4.1 Electromagnetic shielding/absorption

With the extensive utilisation of electronic devices for civil and military purposes there is a strong desire to reduce electromagnetic (EM) pollution due to its associated negative impact on human health, national defence security and electronic safety.[121,226–231] EM wave shielding, or absorption materials, are designed to address this challenge. Fig. 9 shows a schematic diagram that illustrates the working mechanism of so-called microwave absorption materials (MAMs). When incident waves reach the surface of the material, they are either reflected off the sample or transmitted into a porous channel; where multiple reflections occur and the energy is dissipated. The difference between MAMs and electromagnetic-induced (EMI) shielding materials is that MAMs require EM waves to be transmitted into
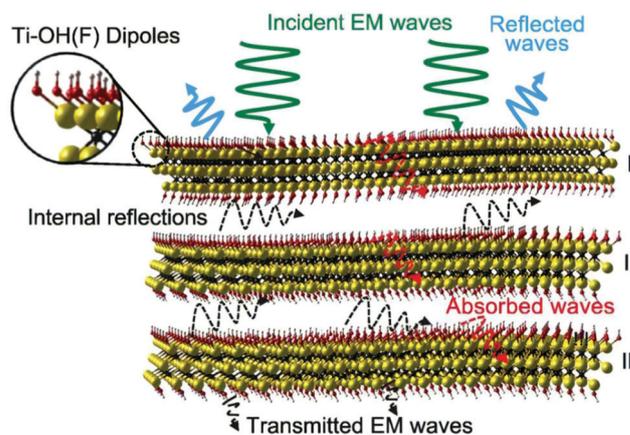

Fig. 9 Electromagnetic wave attenuation through a shielding material.[233]

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the materials to be dissipated rather than simply reflected at the surface.[150,196] MAMs achieve EM attenuation through a combination of magnetic loss, electric loss and geometry (morphology).[226,228] Magnetic loss originates from the eddy current effect, natural ferromagnetic resonance and exchange resonance. For ferromagnetic materials, the magnetic loss ability is related to initial permeability (μ_i), as shown in eqn (1), where a and b represent constants relating to the material being used, and λ, k, ξ, M_s and H_c are the wavelength, magnetisation constant, elastic strain parameter of a crystal, saturation magnetisation and coercivity, respectively. Eqn (1) shows that high M_s and low H_c values are required to increase μ_i, to deliver better magnetic loss ability.[226,232]

$$\mu_i = \frac{M_s^2}{akH_cM_s + b\lambda\xi} \tag{1}$$

For electromagnetic wave absorption (EWA) properties, complex permittivity $\varepsilon_r = \varepsilon' - j\varepsilon''$ and complex permeability $\mu_r = \mu' - j\mu''$ are two critical factors in microwave absorption property determination. The real part (ε' or μ') corresponds to the energy storage whilst the imaginary part (ε'' or μ'') corresponds to the energy dissipation. Dielectric loss and magnetic loss are expressed by $\tan\left(\dfrac{\varepsilon'}{\varepsilon''}\right)$ and $\tan\left(\dfrac{\mu'}{\mu''}\right)$, respectively.[196,226,228]

Reflection loss (RL) is calculated from eqn (2)–(4):[196,226,228]

$$RL(dB) = 20\log\left|\frac{Z_{in} - Z_0}{Z_{in} + Z_0}\right| \tag{2}$$

$$Z_0 = \sqrt{\mu_0/\varepsilon_0} \tag{3}$$

$$Z_{in} = \sqrt{\frac{\mu_r}{\varepsilon_r}}\tanh\left[j\frac{2\pi}{c}\sqrt{\mu_r\varepsilon_r}ft\right] \tag{4}$$

where, Z_{in} and Z_0 are the input impedance of the absorber and impedance of free space, ε_0 and μ_0 are vacuum permittivity and permeability, and ε_r, μ_r, f, c and t are the relative complex permittivity, permeability, frequency of the electromagnetic wave, velocity of electromagnetic waves in free space and thickness of the absorber, respectively.[226] Accordingly, thickness plays an important role in the magnitude of RL.[226,227] RL decreases and the absorption peaks shift to lower frequencies with increasing absorber thickness. Dielectric and magnetic losses also influence RL and are used in combination to improve the microwave absorption performance.[228] Additionally, crosslinking the structure can cause a geometric effect that contributes to RL, as demonstrated with Fe/C cross-linked network structures.[228]

Carbon fibres can be equipped with magnetic properties through the addition of MNPs to create EM absorbing materials. The ultimate EM wave absorption capability is a direct result of both the dielectric and magnetic properties, with higher M_s and lower H_c values also contributing to the magnetic wave absorption properties.[226,228] Notably, NiFe$_2$O$_4$/multi-walled carbon nanotubes (MWCNTs)/carbon nanofibrous membranes, with thicknesses of 2–5 mm, have been created as MAMs that achieve high permeability and reflection losses > 20 dB (5.36–18 GHz).[142]

4.2 Separation: pollutant treatment and catalysis

Functional pollutant absorbers and photocatalysts are used extensively in wastewater treatment. However, the challenge on how to avoid secondary pollution of any residual absorber or catalysts remains a significant problem. One method used to address this issue is the incorporation of electrospun NFs with unique magnetic properties. After degradation, these materials are easily recycled by use of a permanent magnet.[45,158,166,177,199,200,202,234–247] For example, Liang et al.[248] prepared multifunctional switchable chemosensors for Hg^{2+} with fluorescent probe 1-benzoyl-3-[2-(2-allyl-1,3-dioxo-2,3-dihydro-1H-benzo[de]isoquinolin-6-ylamino)-ethyl]-thiourea (BNPTU), PNIPAM copolymer moiety for thermo-responsiveness and on–off photoluminescence and Fe$_3$O$_4$ NPs for easy removal. Chang et al. deposited bismuth oxyiodide nanoflakes on α-Fe$_2$O$_3$ NFs via the SILAR (successive ion layer absorption and reaction) method.[237] The final product reached a maximum specific surface area of 35.27 m^2 g^{-1}, rhodamine B degradation efficiency of 98% after 2 hours of visible light irradiation and an M_s value of 0.9 emu g^{-1}, demonstrating its potential for use as practical separable photocatalysts. As these MNPs render the fibres ferromagnetic, or superparamagnetic, they can be attracted to external magnetic fields.[239,249–251] In addition to this, it has been shown that under stronger magnetic fields the catalysts can be demulsified and therefore recycled back for reuse.[252]

4.3 Tissue engineering scaffolds, cell culture and differentiation

Electrospun NFs have been extensively explored as scaffolds for regenerative medicine. Recently, superparamagnetic fibrous scaffolds have been discovered to have a positive effect regarding cell differentiation (Fig. 10). Furthermore, after implanting the scaffold into the human body, the magnetic property enables the device to be manipulated remotely. The most common approach in this area is to combine MNPs, such as γ-Fe$_2$O$_3$,[253–255] Fe$_3$O$_4$,[8,85,256–259] CoFe$_2$O$_4$ and bioglass,[260] with bio-compatible polymers [such as PVA,[253–255,260] polycaprolactone (PCL), poly(L-lactide), poly(lactic-co-glycolic acid) (PLGA), poly(succinimide) and poly(aspartic acid)[8,85,256–259]] to create flexible composite fibrous mats. These mats have been shown to exhibit enhanced mechanical properties compared to pristine non-magnetic NF mats. NFs with a dense, uniform, surface coverage of MNPs are targeted, however, their production remains a challenge.[144,242]
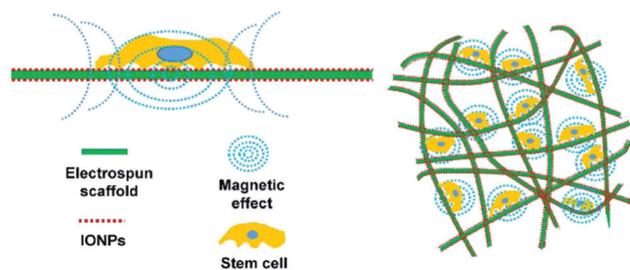

Fig. 10 Schematic showing the mechanism and effect of the IONP-assembled electrospun scaffold on the cells.[144] Adapted with permission from ref. 144. Copyright (2021) American Chemical Society.

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There have been attempts to employ electrospun fibrous templates to create tissue engineering scaffolds, where such methods appear to facilitate better distribution of MNPs on the surface of the NFs. For example, the layer-by-layer assembly technique was used to afford virtually continuous, compact and uniform α-Fe$_2$O$_3$ nanoparticle immobilisation on PLGA/PCL electrospun scaffold surfaces.[144] In this example, the superparamagnetic scaffold was found to significantly enhance the differentiation of adipose-derived stem cells for the treatment of osteogenesis. Moreover, the capping layer brings about auxiliary benefits such as hydrophilicity, elasticity of the interface and the affinity for stem cells. In other work, composite Fe$_3$O$_4$/silk fibroin NFs were compared to silk fibroin fibrous templates with coated Fe$_3$O$_4$ NPs to show the effect of the fabrication method on the final properties of the fabrics. The composite material (electrospun from a single solution) was more effective as a cell scaffold whilst the fibrous mat coated with MNPs was more suitable as a magnetic-sensitive interface.[8]

4.4 Hyperthermia treatment

The treatment of cancer cells includes chemotherapy, radiation therapy and surgery. Whilst these treatments have been used for decades it has been suggested that methods typically used to treat hypothermia could be an effective and ideal approach. This is based on the incorporation of superparamagnetic NFs which can either give out a fatal amount of heat to kill the cancer cells or the raised temperature of targeted tissues can enhance the effects of other therapy (such as drug release as shown in Fig. 11) used to destroy the cancer.[107,109,145,261,262] Superparamagnetic fibres produce a thermal effect in response to an applied alternating magnetic field. The significant advantage of superparamagnetic NFs is their demagnetisation in the absence of an external field. Non-toxic, biodegradable polymeric matrices[109] serve as carriers to provide mechanical support alongside other performance-enhancing properties, such as inhibiting the overgrowth of malignant tissues.[107] The further merit of NFs in biomedical applications lies with their large surface area-to-volume ratios that lead to faster degradation of the scaffold after they have served their purpose. A large ratio of superparamagnetic NPs relative to polymer is required to provide an adequate number of accessible sites and immobilise the NPs on the NFs.[145] There are reports where the number of MNPs on fibres have been quantified. Chen *et al.* used Fe^{2+} to chelate with alginates before fabricating a crosslinked Fe$_3$O$_4$–SA/PVA mat by chemical co-precipitation. Immobilising the NPs on the fibre was then shown to eliminate cytotoxicity effects towards human lung fibroblast cells.[107,109]

4.5 Drug delivery

Recent developments in nanotechnology have led to polymeric matrices (*e.g.* NFs) being exploited as controllable drug release media.[263] Encapsulating bioactive agents in a polymer matrix is an effective method for preventing drug degradation in potentially hostile environments (*e.g.* high or low pH). Electrospun ultrathin fibres exhibit ideal characteristics such as high porosity, large surface area and diverse controllable morphologies to facilitate the development of electrospun fibrous drug delivery systems.[88,264]

For example, a core–shell fibrous drug delivery system was produced by coaxial electrospinning using Eudragit S100 as the shell and a core composed of PEO loaded with Gd(DTPA) (gadolinium diethylenetriamine pentaacetate hydrate, magnetic resonance contrast agent) and indomethacin (model therapeutic agent).[265] Eudragit is insoluble in acidic environments and when used as oral medication protects the core ingredient as it passes through the stomach. The core materials are only released when the fibres reach the intestinal fluids, enabling this core–shell fibre delivery system to achieve targeted drug delivery to the colon. In another example, PVA/ferritin NF hydrogels with controllable magnetic properties were fabricated by partially unfolding the ferritin protein shell at varied mixing temperatures.[100] The negative image contrast generated by ferritin in the PVA matrix under MRI provided a method for *in vivo* imaging of the tissue-engineered scaffolds.

Meanwhile, magnetic reagents loaded into electrospun fibres have been shown to demonstrate synergistic effects. Sasikala *et al.*[266] designed and synthesised an implantable magnetic nanofibrous device for hyperthermia treatment comprised of iron oxide nanoparticles and tumour-triggered controlled drug release (bortezomib). The fibres exhibited a synergistic anticancer effect by applying the hyperthermia treatment and drug delivery simultaneously. In other work, a similar design was employed by embedding MNPs (as heat generators) and DOX (doxorubicin, anticancer drug) inside a temperature responsive copolymer of poly(*N*-isopropylacrylamide)-*co*-(*N*-hydroxymethylacrylamide) [P(NIPAM-*co*-HMAAm)].[1] When placed in an alternating magnetic field, the crosslinked P(NIPAM-*co*-HMAAm) NF mesh showed
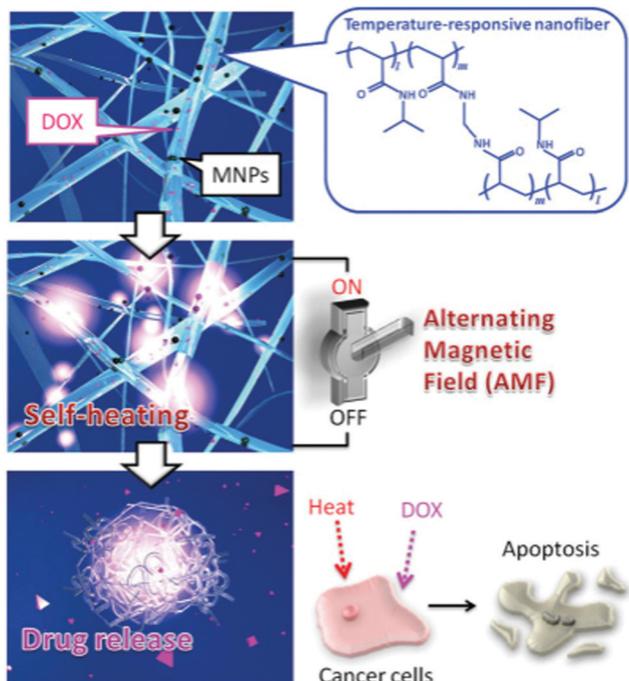

Fig. 11 Design concept for a smart hyperthermia NF system that uses MNPs dispersed in temperature-responsive polymers.[1]

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reversible changes in swelling, which created an 'on–off' response allowing DOX to be released.

Beyond their use in externally triggered controlled release, the addition of MNPs can have a varied effect on drug release rate, depending on how the MNPs impact on the system within which they are placed. Demir et al.[267] found that RhodB (hydrophilic dye) loaded in PCL NFs was released faster when MNPs were present due to the magnetic interaction between the nanoparticles and drug. Similarly, Haroosh et al.[268] also found that the addition of MNPs increased drug release rate. In this case, the inclusion of the MNPs increased the conductivity of the electrospinning solution and also decreased its viscosity, which led to the production of thinner fibres. The larger specific surface area of the thinner fibres resulted in the increased drug release rate observed. Conversely, Wang et al.[18] found that the release rate of indomethacin and aspirin (model drugs) was not affected by the incorporation of Fe_3O_4 NPs within the cellulose matrix; even when the MNPs occupied nearly 50% of the fibre mass.

In short, MNPs can be incorporated into NFs for controlled, triggered drug delivery, or to enhance the properties of the nanofibrous drug delivery system. In the latter case, caution must be made when designing and fabricating the device, as the incorporation of the magnetic nanoparticles can have variable effects on the drug delivery performance of the system.

4.6 Nanogenerators

The development of wearable electronic devices has created a demand for flexible, lightweight, self-powering and energy scavenging technologies.[269,270] In 2001, Glynne-Jones et al.[271] introduced the piezoelectric vibration-powered microgenerator before various materials, such as organic materials, metals, textiles and papers were developed for use in nanogenerators.[234,272–276]

Fabricating PVDF fibrous nanogenerators by electrospinning has been widely researched and is considered a good method to enhance energy generation performance.[277–281] Im et al.[282] fabricated Fe_3O_4/PVDF composite NFs for use in a triboelectric nanogenerator device. Incorporating PVDF resulted in an increase in the surface area and preferential formation of the PVDF polar β-phase, which, in turn, enhanced the triboelectric performance of the device. Additionally, increasing the Fe_3O_4 content increased the output voltage initially from 124 to 138 V and enhanced the EMI shielding performance when added in small quantities. However, a decrease in the output voltage (94 V) was then observed due to aggregation when the α-Fe_2O_3 loading was too high (28.3 wt%). Similarly, the tensile strength of the NF initially increased before decreasing, this time due to the dispersion strengthening mechanism. In another example, Wu et al.[283] reported a method to synthesise a lead zirconate titanate textile (using PVP as a scaffold) in which the nanowires were parallel to one another. This nanogenerator generated a 6 V output voltage, 45 nA output current and qualified to power common liquid crystal displays and UV sensors. Additionally, the nanowires were soft and flexible, ideal for use in wearable nanogenerators.

In summary, organic materials have been investigated as potentially useful counterparts in magnetic NFs for use as nanogenerators. As demonstrated, incorporating polymers, such as PVDF, into the NFs has resulted in enhancement to the triboelectric performance of such devices. Further advances have also been made to produce soft and flexible materials to be used in wearable nanogenerator devices.

4.7 Data storage and transfer

It is well known that modern data storage media function by storing information as binary data (marked as 0 and 1). Magnetic materials are widely used for data storage as they can be magnetised into two opposing directions, which can respectively represent the binary codes, 0 and 1. Both simulations and experiments have shown that nanocylinders or beads that are several hundred nanometres in diameter can form a vortex state under small or vanishing external magnetic fields.[284] The magnetisation of these nanostructures rotates in a closed loop and is often referred to as a vortex-core. In a fibre, the magnetisation state of the beads controls the signal to transfer from one side of the beads to another. The vortex-core can be switched between the two orientations using short magnetic field pulses,[285–287] which can be used to write binary data.

Electrospinning is considered a useful tool to create combinations of nanofibrous mats with embedded beads. The beaded fibres can be fabricated when the polymer solid content in the spinning solution is reduced. Döpke et al.[50] electrospun PAN/Fe_3O_4/α-Fe_2O_3/NiO beaded fibres from 14 wt% polymer solution and computer simulated signal transfer in the beaded fibres. Without an applied static magnetic field the signal transferred through the beads in the shape of a snake-like gyrotropic precess from one side to another. Upon applying a static magnetic field, the magnetisation of the bead fully oriented along the direction perpendicular to the fibre and the signal was blocked. This 'on'/'off' state could not only be used for data storage, but also as logic elements (such as AND or NAND) for neuromorphic computing. Blachowicz et al.[288] further simulated the magnetisation reversal mechanisms under different local spatial distributions and mutual influences of neighbouring magnetic fibres. They found a tendency towards larger coercive fields as the NFs were distributed at a larger random angle range. Fibre mats consisting of two types of NFs (with and without branches) were also simulated. In this case the magnetisation reversal was found to start at smaller negative magnetic fields and end at larger negative fields compared to that of the single nanofibre.

In general, using electrospun magnetic fibres for data storage and neuromorphic computing is still in its infancy but demonstrates promising results. Of the work currently completed there has been a focus on simulating magnetic properties with respect to complex intra- and inter-fibre electrospun MNFs. Blachowicz et al.[288] studied the influence of numbers and dimensions of contact points of electrospun NFs, and proposed a scheme to verify the mechanism. However, this application requires further investigation to enable widespread implementation.

5. Summary and future prospects

Current production methods used to make electrospun MNFs can be divided into two categories. The first introduces magnetic

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components into the electrospinning precursor solutions to fabricate continuous MNFs. The second method uses electrospun MNFs as templates in order to process them into various morphologies and structures. Despite the fact that the magnetic properties of MNPs have been widely studied, the overall magnetic properties of resultant MNFs cannot be predicted since the interactions between polymers and MNPs are not well-understood and are therefore rarely discussed. Many types of MNPs have been added to polymeric materials. In some cases, the MNPs and polymers were simply blended and there were no interactions between them, resulting in a linear trend of M_s with MNP content. However, for many composite systems a non-linear trend is observed. M_s values were often increased, and yet some were attenuated, compared to the simple blended systems. However, to the best of our knowledge, there is no evidence provided to establish the relevant interactions between MNPs and polymer materials and the mechanism behind the non-linear relationship remains unclear. Theoretical modelling is required to predict the relationship between structure and magnetic performance for pure inorganic MNFs. Given that this field is very much in its infancy, novel composite and inorganic magnetic systems have been typically investigated to showcase their unique magnetic performance and are not employed in a given application.

Where specific applications are described there are often challenges that remain to be addressed. Loading magnetic resonance imaging contrast agents via electrospinning was a common method to boost the MRI signal of fibre-based materials for in vivo applications. However, passively imaging the materials is not enough for real medical treatment. Smart or environmentally responsive magnetic fibres are needed not only for location and shape information of the fibres under MRI, but also for linking physiological indices, microenvironment parameters (pH, temperature), degrees of degradation or concentration of bio-factors with MRI signal intensity to provide greater internal body information.

Nanofibres with deposited particles have been found to be useful in wastewater treatment due to their high surface area (as compared with MNFs that have MNPs embedded in the matrix), leading to more catalytically active sites. However, MNPs that have been deposited onto NFs (post-electrospinning) are often detached from the fibrous matrix after several absorption–desorption test cycles, which hampers their viability for long-term use. Therefore, further investigation is needed to identify how MNPs could be adhered to the NFs (to survive hundreds of cyclic tests) would be meaningful work. Sensing technologies for heavy metal ions and toxic gases that can cause ill effect to humans have been summarized elsewhere.[251] Integrating existing pollutant removal technologies with advanced nanofibrous systems, alongside in-depth studies on these toxicants would significantly extend their applications.

In summary, advances in electrospinning have allowed a wide range of magnetic nanofibrous materials to be created over the past two decades. These advanced materials have shown huge potential and are set to play a role in smart technologies of the future. To unlock their true potential, significant work is needed to understand the key interactions between the various components in the system to overcome the current drawbacks that have been encountered.

The processing methods of electrospun magnetic composite NFs and pure inorganic NFs from templates, their properties and related applications have been summarised and discussed throughout this review. Alongside this, key areas for future research have been highlighted with the aim of stimulating advances in the development of electrospun magnetic nanomaterials for a wide range of applications.

Glossary

AR	Acid red 27
BA-a	Bisphenol-A, paraformaldehyde and aniline monomer hybrid
$(BH)max$	Maximum magnetic energy product
BM	Biological microscopy
BNPTU	1-Benzoyl-3-[2-(2-allyl-1,3-dioxo-2,3-dihydro-1H-benzo[de]isoquinolin-6-ylamino)ethyl]-thiourea
CA	Cellulose acetate
CMC	Carboxymethyl-cellulose
CMR	Colossal magneto resistance
CNFs	Carbon-based nanofibers
CS	Chitosan
CTMB	Cellulose tris-(4-methylbenzoate)
DCM	Dichloromethane
DMA	N,N-Dimethylacrylamide
DMF	Dimethylformamide
DMS	Diluted magnetic semiconductor
DNA–CTMA	Deoxyribonucleic acid–cetyltrimethyl-ammonium chloride
EMI	Electromagnetic interference
EMS	Electromagnetic interference shielding
EWA	Electromagnetic wave absorption
α-Fe$_2$O$_3$	Hematite
γ-Fe$_2$O$_3$	Maghemite
FESEM	Field emission scanning electron microscopy
Gd(DTPA)	Gd(III) (diethylenetriamine pentaacetate hydrate)
GO	Graphene oxide
H_c	Coercivity
HPMCP	Dehydroxypropyl methyl cellulose phthalate
H_s	Saturation field
IONPs	Iron oxide nanoparticles
MADO	P(MMA-co-DMA)
MAMs	Microwave absorption materials
MEK	Methyl ethyl ketone
MGNPs	Magnetic glass ceramic nanoparticles
MMA	Methyl methacrylate
MNF(s)	Magnetic nanofibre(s)
MNP(s)	Magnetic nanoparticle(s)
M_r	Remanent magnetisation
M_s	Saturation magnetisation
MWCNTs	Multi-walled carbon nanotubes
NFs	Nanofibers

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NPs	Nanoparticles
P(AN-co-AA)	Poly(acrylonitrile-co-acrylic acid)
P123	Pluronic
PA6	Polyamide-6
PAA	Polyamic acid
PAAm	Polyacrylamide
PAN	Polyacrylonitrile
PANI	Polyaniline
PBT	Poly(butylene terephthalate)
PBZ	Polybenzoxazine
PCL	Polycaprolactone
PDA	Polydopamine
PDLLA	Poly(D,L-lactide)
PDMS	Polydimethylsiloxane
PEI	Polyethyleneimine
PEK-C	Phenolphthalein polyetherketone
PEO	Poly(ethylene oxide)
PET	Poly(ethylene terephthalate)
PEtO$_x$	Poly(2-ethyl-2-oxazoline)
PF–Na	Polyfluorene–Na
PHB	Poly(3-hydroxybutyrate)
PHEMA	Poly(2-hydroxyethyl methacrylate)
PHVB	Poly(3-hydroxybutyrate-co-3-hydroxyvalerate)
PI	Polyimide
PLGA	Poly(lactic-co-glycolic acid)
PMMA	Poly(methyl methacrylate)
PNIPAM	Poly(N-isopropylacrylamide)
P(NIPAM-co-HMAAm)	Poly(N-isopropylacrylamide)-co-(N-hydroxymethylacrylamide)
POSS	Polyhedral oligomeric silsesquioxane
PS	Polystyrene
PS-b-PI	Poly(styrene-block-isoprene)
PU	Polyurethane
PVA	Poly(vinyl alcohol)
PVAc	Poly(vinyl acetate)
PVDF	Poly(vinylidene fluoride)
PVDF-TrFE	Poly(vinylidene fluoride-trifluoroethylene)
PVP	Polyvinylpyrrolidone
RL	Reflection loss
RT	Room temperature
SAN	Styrene–acrylonitrile random copolymer
SEM	Scanning electron microscopy
SILAR	Successive ion layer absorption and reaction technique
T_B	Blocking temperature
T_c	Curie (transition) temperature
TEM	Transmission electron microscopy
TEOS	Tetraethyl orthosilicate
THF	Tetrahydrofuran
TPEE	Thermoplastic ester elastomer

Conflicts of interest

The authors declare no competing financial interest.

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