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rsc.li/materials-cElectronic and crystal structure of bi-intercalated titanium diselenide $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ [†]E. G. Shkvarina,^a A. I. Merentsov,^a M. S. Postnikov,^a A. S. Shkvarin,^{a*}
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A comprehensive study of the crystal and electronic structure of TiSe_2 intercalated with two 3d metals, Cu and Ni, is presented. It is found that the intercalation of Cu into Ni_xTiSe_2 at room temperature leads to the displacement of the Ni atoms from the octahedral to tetrahedral sites with respect to the Se sublattice. This effect is explained by the charge transfer from Cu to the TiSe_2 -derived conduction band, which makes the chemical bond of Ni with the lattice more stable. This is because the hybridization between Ni 3d and Se 4p, 4s states is stronger than that between Ni 3d and Ti 3d in mono-intercalated Ni_xTiSe_2 . Therefore, it is possible to control the type of the coordination of the intercalated transition metal atom by changing the concentration of doped electrons.

Introduction

Titanium dichalcogenides TiCh_2 (Ch = S, Se, Te) have been studied for a long time.^{1–6} The intercalation of 3d transition metals into TiCh_2 alters significantly the physical properties and electronic structure of these materials.^{7–10} To date, the effect of intercalation on the electronic structure of layered transition metal dichalcogenides is well understood. However, there is little research on the formation of materials intercalated by several different metals simultaneously. A strong magnetic and, possibly, chemical interaction between intercalated Co and Fe in TiS_2 was reported in ref. 11. It has been shown that the intercalation of 0.5 mol% Fe into $\text{Co}_{0.25}\text{TiS}_2$ changes the character of the magnetic ordering from ferromagnetic (or ferrimagnetic) to antiferromagnetic. Such a small concentration means that the change of the magnetic ordering is due to the interaction between the Fe and Co atoms *via* the host lattice.

On the other hand, an influence of co-intercalation in $(\text{Cr,Ni})_{1/3}\text{TaS}_2$ on the physical properties of the material has not been observed.¹² This difference may be the result of the different electronic structure of the TiS_2 and TaS_2 host lattices. Indeed, the electronic configuration of Ti in TiCh_2 , Ch = S, Se, Te, can be described as d^0 , while Ta in TaCh_2 has the d^1

configuration. This results in the location below the Fermi level of the M states in M_xTaCh_2 and directly at the Fermi level in M_xTiCh_2 . Since it is the states at the Fermi level that are predominantly responsible for the character of the chemical bond, the co-intercalation effect in TiCh_2 should be visible more clearly. Little is known about $\text{Cu}_x\text{Fe}_y\text{TiSe}_2$,^{13,14} although the crystal structure and thermodynamic stability in a wide temperature range were studied for this system. It has been shown that the subsystems of Fe and Cu are independent. $\text{Cu}_x\text{M}_{0.25}\text{TiSe}_2$ (M = Mn, Co, and Ni) co-intercalated compounds were previously investigated by our group.¹⁵ However, the research was focused on the lattice parameters and thermodynamic stability only.

It should be noted that the synthesis of all the above mentioned co-intercalated materials was performed at high temperature. This provides the equilibrium for both the sublattices of intercalated metals and the whole crystal lattice. However, the properties of the co-intercalated TiCh_2 are likely influenced by the charge transfer, mutual influence of the intercalated metals, their interaction with the host lattice and magnetic interaction. In this study, we focus on Ni and Cu co-intercalation in TiSe_2 ; the possibility of co-intercalation of these metals was reported in ref. 15. Since Cu can be intercalated into TiSe_2 at room temperature,¹⁶ this provides minimal distortion of the host lattice and the Ni sublattice, but the possibility of charge transfer is preserved. In addition, both Cu in Cu_xTiSe_2 and Ni in Ni_xTiSe_2 intercalation compounds occupy the same positions in the van der Waals gap, which are octahedrally coordinated by chalcogen. Therefore, a competition for the occupation of these positions can be expected.

The crystal structure of the TiSe_2 -based intercalation compound is shown in Fig. 1. Two types of positions in the

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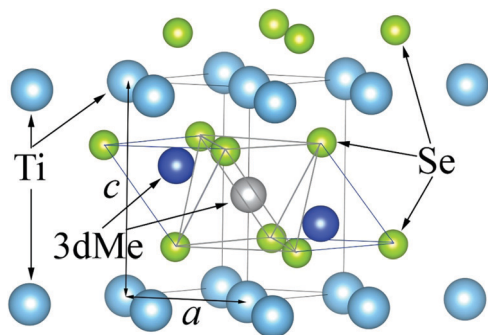


Fig. 1 Fragment of $M_x\text{TiSe}_2$ crystal structure. The tetrahedral sites are shown as two blue spheres and the octahedral site is shown as a gray sphere.

TiSe_2 interlayer space are available for intercalated metals with respect to the Se sublattice: octahedral (shown in gray) and tetrahedral (shown in blue) sites. In all TiSe_2 -based compounds intercalated by noble and transition metals, intercalated atoms are located in octahedral sites.

This study is focused on the synthesis and systematic study of the crystal and electronic structures of the co-intercalated $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ compound to clarify the mutual influence of the intercalated metals.

Experiment

The polycrystalline phase of $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ was synthesized by Ni intercalation into TiSe_2 followed by Cu intercalation into Ni_yTiSe_2 . The synthesis procedures for TiSe_2 as well as for Ni intercalation into TiSe_2 are described in detail elsewhere.¹⁵ Cu intercalation was performed at room temperature by the method described in ref. 16. The $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ and $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$ systems (see Cu concentrations x in Tables 1–3) were successfully synthesized. An increase in the Ni concentration to 50 mol% led to a crumbling of the samples into powder. The contact area between grains became smaller and, consequently, the homogeneous state was not achieved. The only exceptions were the $\text{Cu}_{0.1}\text{Ni}_{0.5}\text{TiSe}_2$ and $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{TiSe}_2$ compounds.

The homogeneity of the samples was checked by X-ray diffraction analysis. Crystal structure and phase composition

were determined using X-ray powder diffraction on a Shimadzu XRD 7000 Maxima diffractometer ($\text{Cu K}\alpha_1$ radiation, graphite monochromator, $2\theta = 12\text{--}92$, $\Delta\theta = 0.02$, exposure time 1.2 s). The diffraction patterns contained only peaks of the main phase and those of metallic copper (which is not surprising, because the synthesis was carried out at room temperature). The crystal structure refinement was performed using GSAS (General Structure Analysis System) software.¹⁷ The chemical compositions of the obtained intercalation compounds are summarized in Tables 1–3.

The single crystals for the soft X-ray spectroscopy measurements were grown from the powdered $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ materials using the gas transport reaction technique in evacuated quartz ampoules with I_2 as the gas carrier.¹⁸ The single crystals had a shape of thin plates approximately 2–3 mm large and 0.05 mm thick. Two crystals with the composition of $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ and $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ were grown for this work. In addition to the co-intercalated crystals, mono-intercalated Ni_xTiSe_2 ($x = 0.14, 0.46$) and Cu_xTiSe_2 ($x = 0.09, 0.33$) single crystals were investigated too. Details of their synthesis have been reported elsewhere.^{19,20}

The synchrotron radiation X-ray photoelectron (XPS), X-ray absorption (XAS) and resonant X-ray photoelectron (ResPES) spectra for $\text{Ni}_{0.14}\text{TiSe}_2$, $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ and $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ were obtained on the CNR IOM BACH beamline²¹ at the Elettra synchrotron facility (Trieste, Italy). Linearly-polarized X-rays with a polarization vector 60° off the surface plane were used. All the samples were cleaved *in situ* at room temperature in a vacuum chamber with a base pressure lower than 1×10^{-9} Torr. The purity of the surface was confirmed by the absence of oxygen and carbon peaks in the XPS survey spectra. XPS spectra were acquired using a VG-Scienta R3000 hemispherical analyzer at an angle of 60° with respect to the X-ray incidence beam. The Ti 2p, Se 3d, Ni 2p and Cu 2p core levels were measured at normal emission angle using 1114 eV excitation energy. The total energy resolution for the core level and valence band spectra was 0.4 eV. The binding energies were calibrated with respect to the Au 4f_{7/2} ($E_B = 84.0$ eV) signal from a gold reference sample. ResPES spectra were acquired by measuring sets of valence band spectra at photon energies corresponding to the resonances at the Ti L₃ and Ni L₃ edges. Ni and Ti L_{3,2} XAS spectra were acquired in total electron yield mode. The photon energy resolution was

Table 1 Results of the Rietveld refinements for $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$. In the first column the preset Cu concentration along with the Cu concentration obtained from the refinement result (in brackets) are listed

x	a , Å	c , Å	Z Se	Frac. Ni (0 0 1/2)	Frac. Ni (1/3 2/3 z)	Z Ni (1/3 2/3 z)	Frac. Cu (0 0 1/2)	Frac. Cu (1/3 2/3 z)	Z Cu	R_F^2 , %	χ^2
0.05 (0.011)	3.540(1)	5.960(1)	0.2613(2)	0.1	0	—	0.011	0	—	6.36	2.515
0.1 (0.069)	3.541(1)	5.961(1)	0.2605(2)	0.047(4)	0.027(2)	0.47(1)	0.069	0	—	—	—
0.15 (0.108)	3.541(1)	5.961(2)	0.2596(2)	0.009(1)	0.045(2)	0.456(6)	0.108	0	—	3.01	2.528
0.2 (0.128)	3.541(1)	5.961(1)	0.2598(2)	0.003(4)	0.048(2)	0.468(6)	0.122(3)	0.003(1)	0.64(6)	4.34	2.815
0.25 (0.184)	3.540(1)	5.961(1)	0.2587(2)	0	0.05	0.430(5)	0.127(4)	0.028(2)	0.496(9)	2.72	2.420
0.3 (0.260)	3.542(1)	5.962(1)	0.2558(2)	0	0.05	0.435(5)	0.150(3)	0.055(2)	0.613(3)	4.40	2.834
0.35 (0.270)	3.543(1)	5.962(1)	0.2558(2)	0	0.05	0.399(5)	0.162(3)	0.054(2)	0.623(3)	5.70	—
0.4 (0.307)	3.543(1)	5.962(1)	0.2530(2)	0	0.05	0.367(6)	0.163(3)	0.072(2)	0.652(3)	8.57	2.817
0.45 (0.341)	3.542(1)	5.961(1)	0.2522(3)	0	0.05	0.420(6)	0.185(3)	0.078(2)	0.654(3)	7.99	3.476
0.5 (0.303)	3.540(1)	5.963(1)	0.2556(2)	0	0.05	0.402(6)	0.169(4)	0.067(2)	0.630(3)	6.15	3.035



Table 2 Results of the Rietveld refinements for $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$. In the first column the preset Cu concentration along with the Cu concentration obtained from the refinement result (in brackets) are listed

x	a , Å	c , Å	Z Se	Frac. Ni (0 0 1/2)	Frac. Ni (1/3 2/3 z)	Z Ni (1/3 2/3 z)	Frac. Cu (0 0 1/2)	Frac. Cu (1/3 2/3 z)	Z Cu	R_F^2 , %	χ^2
0.05 (0.05)	3.548(1)	5.922(1)	0.2601(2)	0.210(4)	0.020(2)	0.47(1)	0.05	0	—	3.98	2.889
0.1 (0.1)	3.548(1)	5.922(1)	0.2592(2)	0.152(4)	0.050(2)	0.437(5)	0.1	0	—	4.01	2.434
0.2 (0.173)	3.549(1)	5.923(1)	0.2592(2)	0.085(4)	0.083(2)	0.442(3)	0.159(3)	0.007(2)	0.62(2)	3.84	
0.3 (0.258)	3.549(1)	5.922(1)	0.2574(2)	0.119(4)	0.065(2)	0.453(4)	0.166(3)	0.046(2)	0.637(4)	5.62	2.768
0.35 (0.273)	3.549(1)	5.922(1)	0.2562(3)	0.072(4)	0.089(2)	0.435(3)	0.181(4)	0.046(2)	0.658(5)	5.45	3.045
0.4 (0.365)	3.549(1)	5.923(1)	0.2519(2)	0.045(4)	0.103(2)	0.396(2)	0.245(3)	0.060(2)	0.639(3)	7.07	3.327
0.45 (0.399)	3.549(1)	5.921(1)	0.2513(2)	0	0.125	0.361(3)	0.317(4)	0.041(2)	0.636(4)	5.54	2.600
0.5 (0.440)	3.549(1)	5.923(1)	0.2440(2)	0	0.125	0.405(2)	0.365(4)	0.038(2)	0.627(5)	7.45	3.507

Table 3 Results of the Rietveld refinements for $\text{Cu}_x\text{Ni}_{0.5}\text{TiSe}_2$

x	a , Å	c , Å	Z Se	Frac. Ni (0 0 1/2)	Frac. Ni (1/3 2/3 z)	Z Ni (1/3 2/3 z)	Frac. Cu (0 0 1/2)	Frac. Cu (1/3 2/3 z)	Z Cu	R_F^2 , %	χ^2
0.1 (0.1)	3.566(1)	5.928(1)	0.2600(2)	0.377(4)	0.061(2)	0.429(3)	0.1	0	—	4.47	2.587
0.5 (0.5)	3.566(1)	5.929(1)	0.2438(2)	0.151(5)	0.175(3)	0.471(1)	0.5	0	—	6.55	2.379

set to 0.06 eV at Ti L-edge and 0.2 eV at Cu and Ni L-edge, respectively. The photon energy scale was calibrated with an accuracy of 0.1 eV using Au 4f XPS spectra measured with the fundamental and second harmonic of the monochromator. The uncertainty in Ni L-edge calibration is 0.2–0.3 eV.

The XPS and XAS spectra for Cu_xTiSe_2 ($x = 0.09, 0.33$) were obtained on the Russian–German beamline at the BESSY II synchrotron (Berlin, Germany). The Ti 2p, Se 3d and Cu 2p core levels were measured at normal emission angle at 800 eV excitation energy. The total resolution was approximately 0.2 eV. The binding energies were calibrated with respect to the Pt 4f_{7/2} ($E_B = 71.2$ eV) signal from a platinum reference sample. All measurements were performed at room temperature.

The XPS and XAS spectra for $\text{Ni}_{0.46}\text{TiSe}_2$ sample were obtained on the CiPo beamline at the Elettra synchrotron facility. All measurements were performed at room temperature.

The fitting procedure for the XPS spectra was performed using KolXPD software.²²

Results

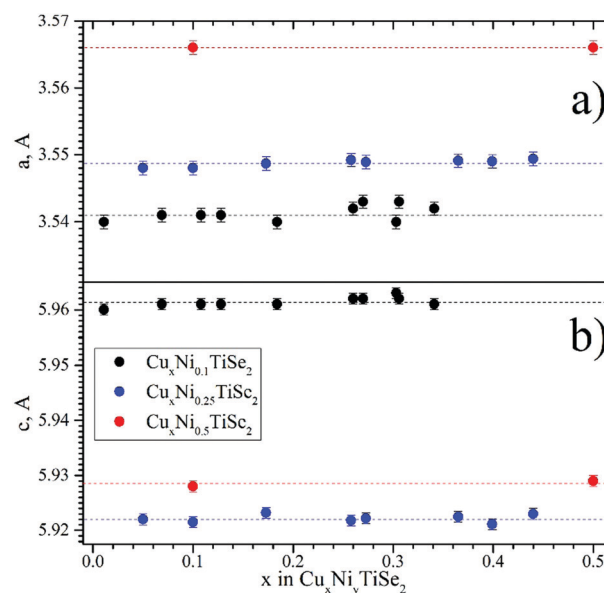
Crystal structure

Following the approach introduced in ref. 15 the crystal structure of $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ was refined in the space group $P\bar{3}m1$ (No. 164), atomic positions Ti(1a) (0 0 0), Se(2d) (1/3 2/3 z), Cu1(1b) (0 0 1/2) (octahedral sites), Cu2(2d) (1/3 2/3 z) (tetrahedral sites), Ni1(1b) (0 0 1/2) (octahedral sites), Ni2(2d) (1/3 2/3 z) (tetrahedral sites).²³ Site occupancy fractions (SOF) for Ni (1b) and Ni (2d) were constrained by total Ni concentration, considering the multiplicity of the positions. The same constraint was used for SOFs for Cu (1b) and Cu (2d). The background was approximated by Chebyshev polynomials with 16 coefficients.

Some X-ray diffraction patterns contained peaks from metallic copper besides the peaks from the main phase. To take this into account, the metallic copper was included into the crystal structure refinement. The fraction of this phase, determined from the

refinement, was used to define real Cu content in the intercalation phase (these values are given in brackets in x columns, Tables 1–3). After that, this Cu content in the intercalation phase was fixed and distributed over the 1b and 2d Cu sites. Therefore, when Cu occupied only one of these sites, the corresponding SOF was taken from the weight fraction for the metallic Cu phase and the error bars for the Cu 1b SOF in these cases are not given. The lattice parameters a and c change only slightly when Cu is intercalated into Ni_yTiSe_2 (see Fig. 2 and Tables 1–3). The changes induced by Cu intercalation into Ni_yTiSe_2 are 2 orders of magnitude weaker than that determined by the same amount of Ni.

Fig. 3 shows the distribution of the Cu and Ni atoms over the octahedral (1b) and tetrahedral (2d) sites in the interlayer space

**Fig. 2** Lattice parameters (a – panel a; c – panel b) for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ obtained from X-ray powder diffraction.

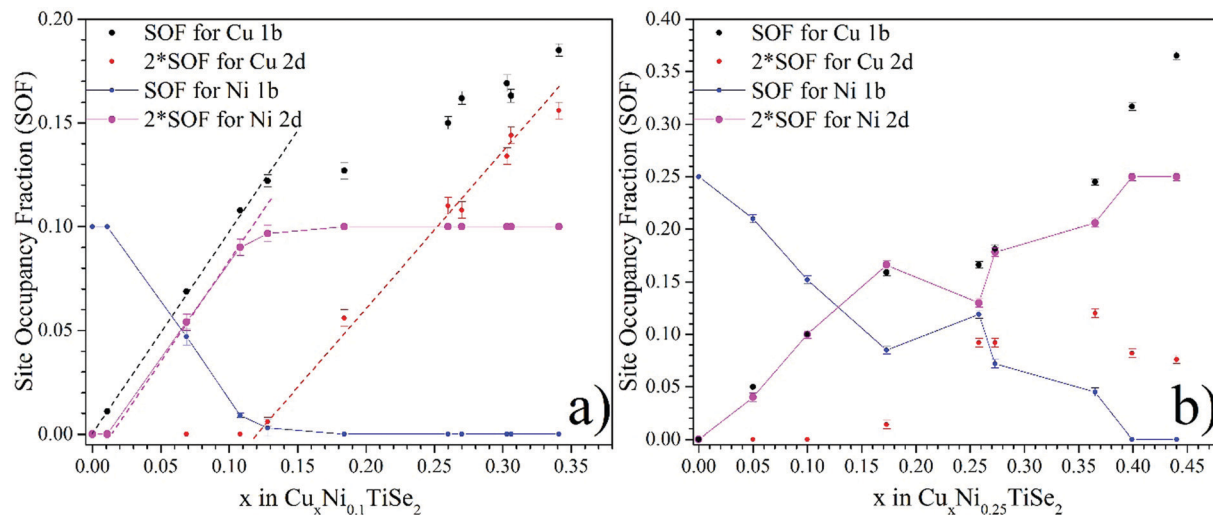


Fig. 3 Occupation of the octahedral and tetrahedral sites (1b – octahedral sites, 2d – tetrahedral sites) in the interlayer space by the Ni and Cu atoms for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ ($y = 0.1$ – panel a, $y = 0.25$ – panel b).

for $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ (panel a) and $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$ (panel b). Since the lattice parameters change slightly, it is expected that the main changes in the crystal structure will be associated with a change in the type of crystallographic sites occupied by the intercalated atoms.

In both $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$ and $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$, the Cu intercalation leads to the displacement of the Ni atoms from the octahedral to tetrahedral sites. However, for $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ compounds in the Cu concentration range of $0 < x < 0.013$ Cu intercalation does not change the Ni coordination (see Fig. 3, panel a). Therefore, it can be assumed that for $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ the interaction between Ni and Cu atoms is negligible in the Cu concentration range of $0 < x < 0.013$. For Cu concentrations of $0.013 < x < 0.12$ in $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ and $0 < x < 0.17$ in $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$, almost all Cu atoms occupy octahedral sites and each Cu atom displaces one Ni atom from the octahedral to tetrahedral site (Fig. 3, panel b).

The diffraction pattern for $\text{Ni}_{0.5}\text{TiSe}_2$ contains extra peaks indicating the monoclinic $I2/m$ ordering, as it has been observed previously.²⁴ However, even 10 mol% Cu in $\text{Cu}_{0.1}\text{Ni}_{0.5}\text{TiSe}_2$ turns the crystal structure to the trigonal $P\bar{3}m1$ one (see ESI†). Obviously, the disordering of the intercalated atoms is caused by the displacement of the Ni atoms in the tetrahedral sites, while monoclinic ordering requires the occupation of just the octahedral sites.

The Cu intercalation leads to a linear increase in the total occupation of the tetrahedral sites with both Cu and Ni atoms (Fig. 4). At the $\text{Cu}_{0.33}\text{Ni}_{0.25}\text{TiSe}_2$ composition, this value reaches saturation and corresponds to the total occupation of the tetrahedral sites, which is equal to 0.33 (considering the multiplicity 2×0.165). This means that one-third of the unit cells contains one intercalated atom in the tetrahedral site. In its turn, this corresponds to the maximum occupation of the tetrahedral sites by both Cu and Ni atoms. Moreover, in the Cu concentration range of $0.05 \leq x \leq 0.15$ in $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$ the tetrahedral sites are initially occupied by the Ni atoms only

but starting from $x = 0.25$ the Cu atoms begin to occupy the tetrahedral sites too. In the Cu concentration range of $0.25 \leq x \leq 0.45$ Cu SOF for the tetrahedral sites does not change significantly, but the Ni atoms are gradually and at $x = 0.45$ completely displaced to the tetrahedral sites (Fig. 4, panel b).

For the $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ system, the total occupation of the tetrahedral sites also increases, but the saturation is not achieved, because the total concentration of the intercalated atoms (Cu + Ni) is less than the saturation value of $x + y = 0.33$ (see Fig. 4, panel a). But in this case the tetrahedral sites are initially occupied only by the Ni atoms until their complete displacement by the Cu atoms. At $x > 0.13$ the Cu atoms begin to occupy the tetrahedral sites too. The data for the $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{TiSe}_2$ sample confirms this conclusion: the tetrahedral sites, in this case, are occupied by Ni up to the concentration of $y = 0.35$, see Table 3 and Fig. 4, panel b.

The total occupation of the octahedral sites (Fig. 5) with the increase in Cu concentration in both $\text{Cu}_x\text{Ni}_{0.1}\text{TiSe}_2$ and $\text{Cu}_x\text{Ni}_{0.25}\text{TiSe}_2$ systems remains constant and approximately equals to the occupation of the octahedral sites by the Ni atoms in $\text{Ni}_{0.1}\text{TiSe}_2$ and $\text{Ni}_{0.25}\text{TiSe}_2$. This is probably due to the minimization of the energy of elastic deformation, because the co-intercalation affects the lattice parameters only minimally.

Both Ni and Cu atoms can occupy the tetrahedral sites, therefore one can expect the same position of these atoms in the Se tetrahedron, because the ionic radii of Cu and Ni are nearly the same. However, their different electronic configuration makes them to have different z coordinates in the Se tetrahedron. In both systems ($y = 0.1$ and $y = 0.25$), the Cu atoms are located closer to the base of the tetrahedron (the face parallel to the (001) plane) ($z \approx 0.62$) and the Ni atoms are closer to the opposite vertex ($z \approx 0.42$) (Tables 1–3).

XPS

Core level XPS provides insight into the charge state of the atoms. In addition, XPS can be used as a homogeneity test for



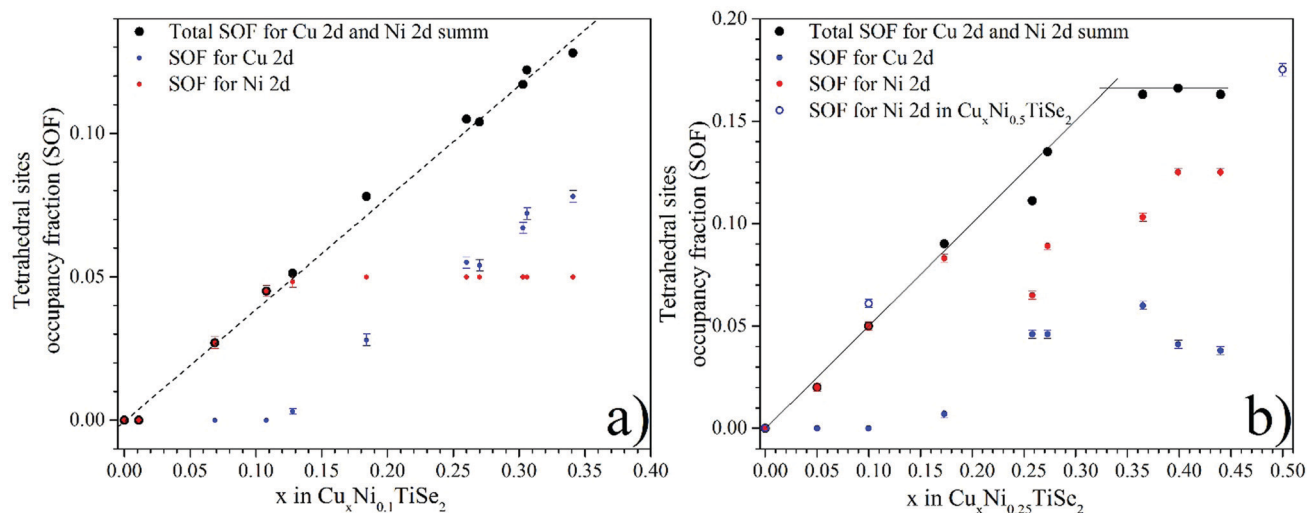


Fig. 4 Occupation of the tetrahedral (2d) sites in the interlayer space by the Ni and Cu atoms for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ ($y = 0.1$ – panel a, $y = 0.25$ – panel b).

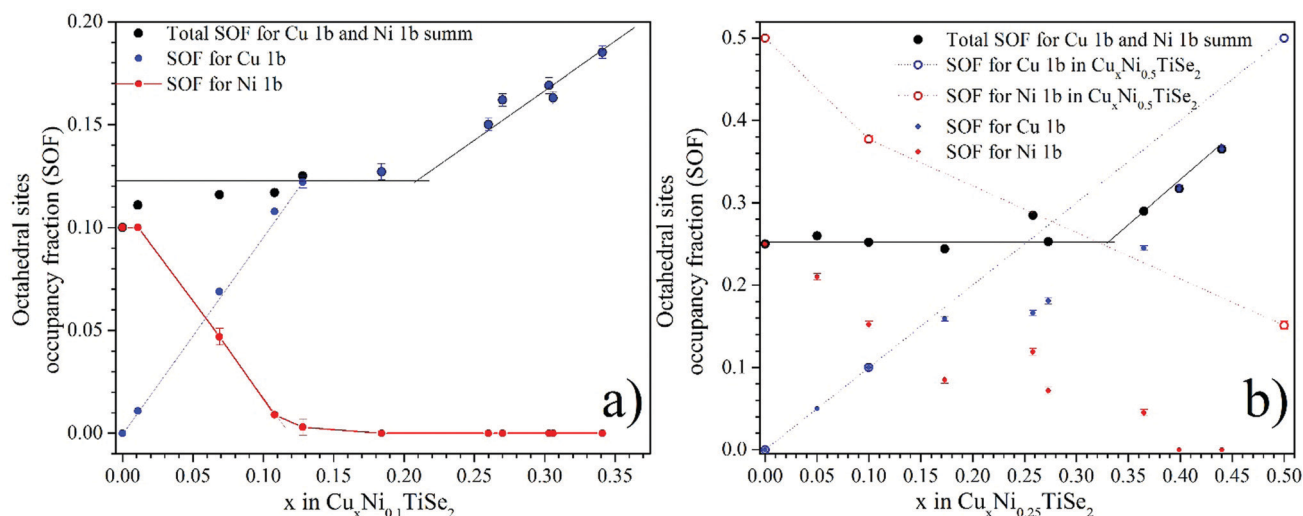


Fig. 5 Occupation of the octahedral (1b) sites in the interlayer space by Ni and Cu for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ ($y = 0.1$ – panel a, $y = 0.25$ – panel b).

the studied samples. The Se 3d spectra (Fig. 6) are specially sensitive to the changes in phase composition and surface quality.^{20,25}

Even at low Ni concentration ($x = 0.14$), a shift towards higher binding energies is observed (see Table 4). According to ref. 19, two Ni concentration ranges can be distinguished in Ni_xTiSe_2 , in which the Se 3d binding energy slightly changes: low Ni concentrations (at $x < 0.25$ the Ni atoms can be considered as isolated ones), and high Ni concentrations (above $x = 0.25$, which corresponds to the percolation threshold of the Ti atoms coordinated by Ni). As it can be seen in Fig. 6 and Table 4, the Cu intercalation into Ni_yTiSe_2 leads to a further shift of the Se 3d line toward the higher binding energies. This is in good agreement with previously obtained data for the Cu_xTiSe_2 system for $x < 0.2$, where Cu intercalation also did not affect the shift of the Se 3d line.²⁰ Probably, the increase in the Se 3d binding energy is not the result of an electron transfer

from Cu to the lattice, but the result of redistribution of the Ni atoms due to the Cu intercalation.

The Se 3d line width is sensitive to the distribution of the Ni atoms over the octahedral and tetrahedral sites. Indeed, for $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$, which is close to $\text{Cu}_{0.1}\text{Ni}_{0.5}\text{TiSe}_2$, almost all the Ni atoms are in the octahedral sites, and the Se 3d line is just slightly broader than that for $\text{Ni}_{0.46}\text{TiSe}_2$. In its turn, in $\text{Ni}_{0.46}\text{TiSe}_2$ all of the Ni atoms occupy only the octahedral sites, whereas in $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ the Ni atoms are nearly equally distributed between the octahedral and tetrahedral sites (see Fig. 3, panel a), and the Se 3d line becomes broad and its double-peak structure cannot be observed. Considering the site occupancy fractions of the octa- and tetrahedral sites (see Table 2, SOF = 0.244 for 1b sites, SOF = 0.173 for 2d sites) we can assume that in $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ the Se 3d states with higher binding (dark green dashed line, Fig. 6) are due to the Se atoms forming the octahedrons, while those with lower binding



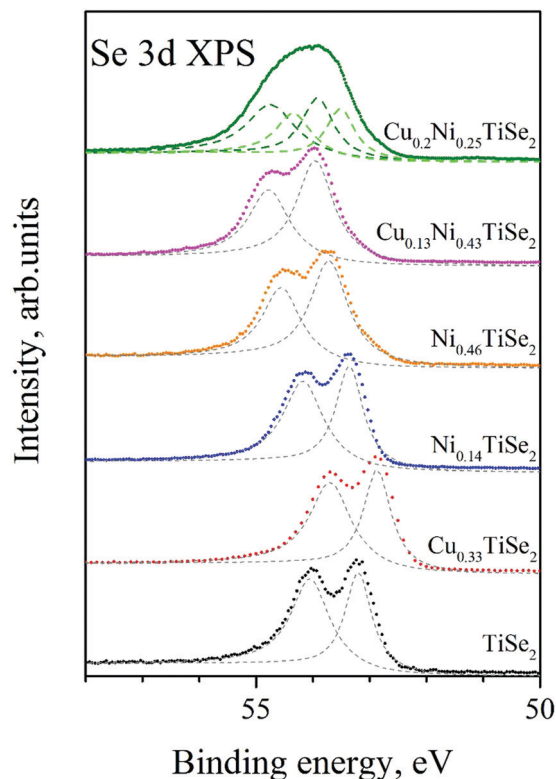


Fig. 6 The Se 3d spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$. The Se 3d spectra for mono-intercalated Cu_xTiSe_2 and Ni_yTiSe_2 are shown as well.

energies (light green dashed line, Fig. 6) are due to the Se atoms forming the tetrahedrons.

The Ti 2p spectra are shown in Fig. 7. They are similar to those for the Ni_yTiSe_2 system.¹⁹ The second (with higher binding energy) Ti state appears due to the presence or absence of intercalated atoms in the second coordination sphere of Ti atoms. This is in a good agreement with the fact that the change in the Cu concentration in Cu_xTiSe_2 did not affect the Ti 2p spectra.²⁰

The Cu 2p spectra shown in Fig. 8 (left panel) are almost identical to those for the Cu_xTiSe_2 system²⁰ and correspond to the Cu^0 or Cu^{I} charge state. This is not surprising since the Cu atoms occupy identical octahedral sites in all of these compounds.

The binding energy of the Ni 2p state (Fig. 8, right panel) does not depend on the Ni or Cu concentration (Table 4). However, a change in the shape of the Ni 2p spectrum is observed. With the increase in the Cu concentration, the intensity of a satellite (marked as Sh in Fig. 8) significantly decreases.

XAS

The X-ray absorption spectra provide information regarding the ratio of the ionic and covalent contribution to the chemical bond of the selected atom.²⁶ This characteristic is related to the local arrangement of the selected atoms and their nearest neighbors; therefore, it should be sensitive to the crystal field effects.

The Ti $\text{L}_{2,3}$ absorption spectra are shown in Fig. 9, left panel. A change in the Cu concentration does not affect the shape and energy position of Ti $\text{L}_{2,3}$ XAS, as in the case of the Ti 2p core-level spectra. The Ti $\text{L}_{2,3}$ absorption spectra can be divided into two groups: the first group for the compounds with low Ni concentration ($x < 0.25$), the second one for the compounds with high Ni concentration ($x \geq 0.25$). The difference in the Ti coordination in these groups is related to the different influence of the neighboring Se atoms. In $\text{Ni}_{0.46}\text{TiSe}_2$ the ordering of Ni resulting in the monoclinic structure is observed. This distorts the Se sublattice and, consequently, leads to a change in the coordination of the Ti atoms by the Se ones.²⁴ Another way to distort the Se sublattice is redistribution of the Ni atoms between the octa- and tetrahedral sites caused by the Cu intercalation. This distortion can change the local crystal field around Ti atoms and influence the lineshape of Ti $\text{L}_{2,3}$ XAS.

The Cu $\text{L}_{2,3}$ absorption spectra are shown in Fig. 9, right panel. As one can see, Cu $\text{L}_{2,3}$ XAS for $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ is almost identical to that for $\text{Cu}_{0.33}\text{TiSe}_2$ without Ni intercalation. As the Cu concentration increases ($\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$), the lineshape changes; additional bands marked with arrows in Fig. 9 appear. In $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$, which is close in chemical composition to $\text{Cu}_{0.1}\text{Ni}_{0.5}\text{TiSe}_2$, all of the Cu atoms occupy the octahedral sites, as in the Cu_xTiSe_2 compounds. Although in $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ the Cu atoms also mostly occupy the octahedral sites, some of them occupy the tetrahedral sites. Therefore, these additional XAS features can be associated with the Cu atoms in the tetrahedral sites.

The Ni $\text{L}_{2,3}$ absorption spectra (Fig. 10) substantially depend on the Cu concentration. Cu intercalation into Ni_yTiSe_2 changes

Table 4 Results of the fitting of XPS spectra. Se $3d_{5/2}$, Ti $2p_{3/2}$ and Ni $2p_{3/2}$ – the energy positions of the corresponding core levels. Ti $\text{L}_3(\text{A})$ and Ti $\text{L}_3(\text{B})$ – the energy position of the A and B peaks in the absorption spectra (Fig. 9). VB_T and VB_R – the energy positions of the T and R peaks in the valence band spectra (Fig. 11), I_T and I_R – intensities of the T and R peaks (Fig. 11)

Sample	Se $3d_{5/2}$, eV	Ti $2p_{3/2}$, eV	Ni $2p_{3/2}$, eV	Ti $\text{L}_3(\text{A})$, eV	Ti $\text{L}_3(\text{B})$, eV	VB_T , eV	VB_R , eV	I_T	I_R
TiSe_2	53.2	455.63	n/a	455.93	457.87	n/a	n/a	n/a	n/a
$\text{Cu}_{0.33}\text{TiSe}_2$	52.87	455.2	n/a	455.97	457.94	n/a	n/a	n/a	n/a
$\text{Ni}_{0.14}\text{TiSe}_2$	53.34	455.67	852.32	455.94	457.95	0.95	0.15	0.18 (0.11)	0.06
$\text{Ni}_{0.46}\text{TiSe}_2$	53.71	455.7	852.38	455.42	457.48	1.2 ¹⁹	0.45 ¹⁹	0.55 ¹⁹	0.24 ¹⁹
		454.64							
$\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$	53.95	456.0	852.44	455.6	457.62	1.1	0.25	0.44 (0.46)	0.18
		454.8							
$\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$	53.51	456.17	852.41	455.6	457.49	1.13	0.3	0.4 (0.56)	0.15
	53.91	454.8							



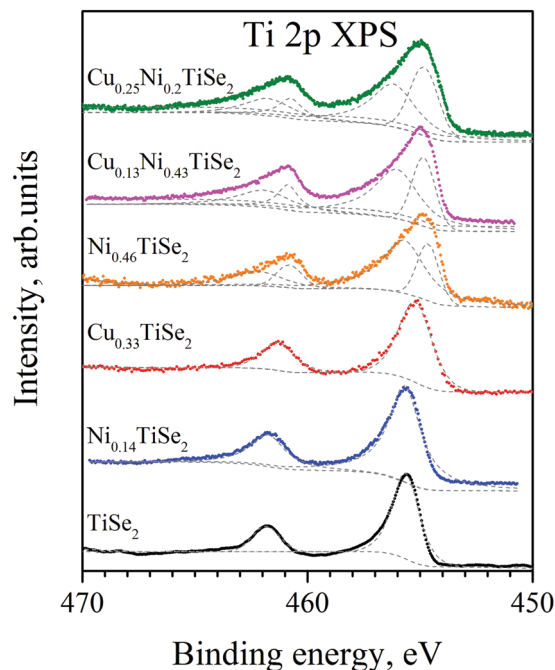


Fig. 7 The Ti 2p spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$. The Ti 2p spectra for mono-intercalated Cu_xTiSe_2 and Ni_yTiSe_2 are shown as well.

the lineshape. The Ni atoms in $\text{Ni}_{0.14}\text{TiSe}_2$ occupy the octahedral sites only. In $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ ($\text{Cu}_{0.1}\text{Ni}_{0.5}\text{TiSe}_2$), the Ni atoms occupy mainly the octahedral sites, whereas in $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ they are equally distributed over the octahedral and tetrahedral sites. Two different coordination of the Ni atoms result in a broadening of the Ni $L_{2,3}$ absorption spectrum for $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ (FWHM = 1.7 eV) as compared to that for $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$

(FWHM = 1.5 eV) (see the inset in Fig. 10). Moreover, there is a pronounced difference between the shapes of the Ni $L_{3,2}$ XAS spectra for $\text{Ni}_{0.14}\text{TiSe}_2$ and $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$. This difference can be related to the different crystal field around Ni atoms in the tetrahedral sites.

Valence band

The photoemission spectra of the valence band (VB) provide insight into the state of the valence electrons that are responsible for the chemical bonds in the material. The use of excitation radiation with a resonant energy close to the 2p–3d transition energy allows for determination of the partial atomic contributions in the density of states in the valence band. The VB spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ are shown in Fig. 11.

The VB spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ differ from those for TiSe_2 by the presence of T and R peaks near the Fermi level. Both of these peaks are typical for the Ni_xTiSe_2 system.¹⁹ One can see that the intensity of the T peak increases with Ni concentration. The intensity of the R peak also increases with Ni concentration but does not depend on the Cu concentration (Table 4). However, for all the studied compounds Cu_xMeCh_2 (Me = Ti, Zr; Ch = S, Se, Te) Cu states are not observed near the Fermi level.^{27,28} To clarify the partial contribution to the states near the Fermi level, the resonant photoemission spectroscopy was used.^{29–32} The valence band spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ obtained in the Ti $2p_{3/2}$ –3d resonant excitation mode are shown in Fig. 12.

One can see that the Cu intercalation into Ni_yTiSe_2 leads to an increase in the intensity of the resonant line directly at the Fermi level. This effect can be explained as a result of the filling of the Ti 3d conduction band. We can conclude that the Cu intercalation in both TiSe_2 and Ni_yTiSe_2 leads to the charge

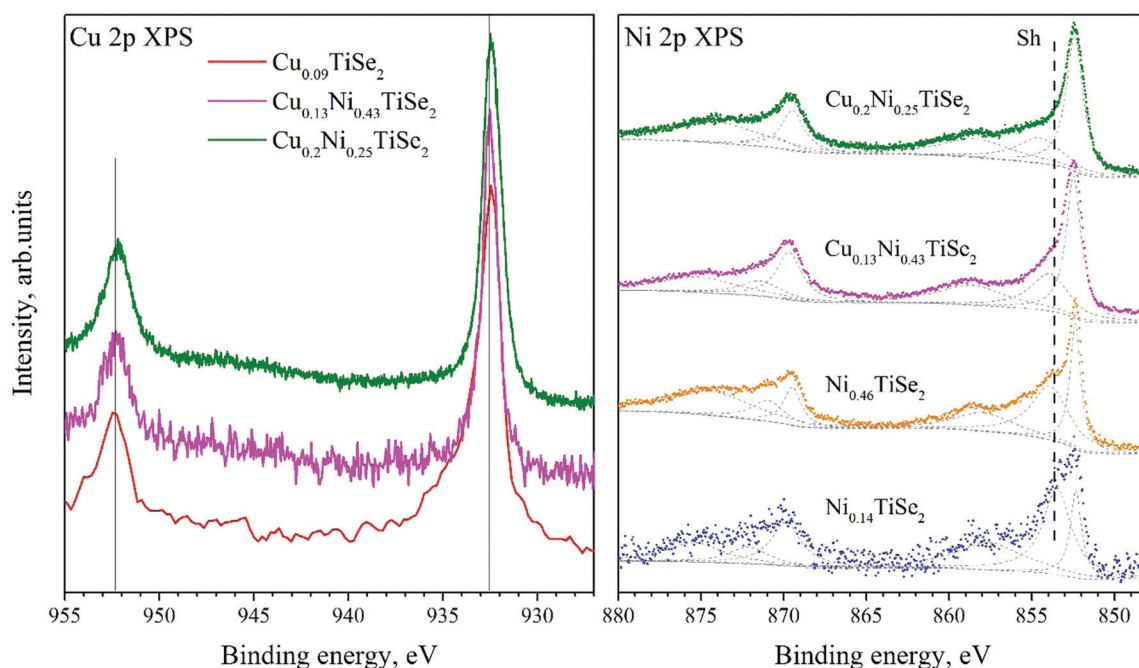


Fig. 8 The Cu 2p spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ and $\text{Cu}_{0.09}\text{TiSe}_2$ (left panel). The Ni 2p spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ and Ni_yTiSe_2 (right panel).



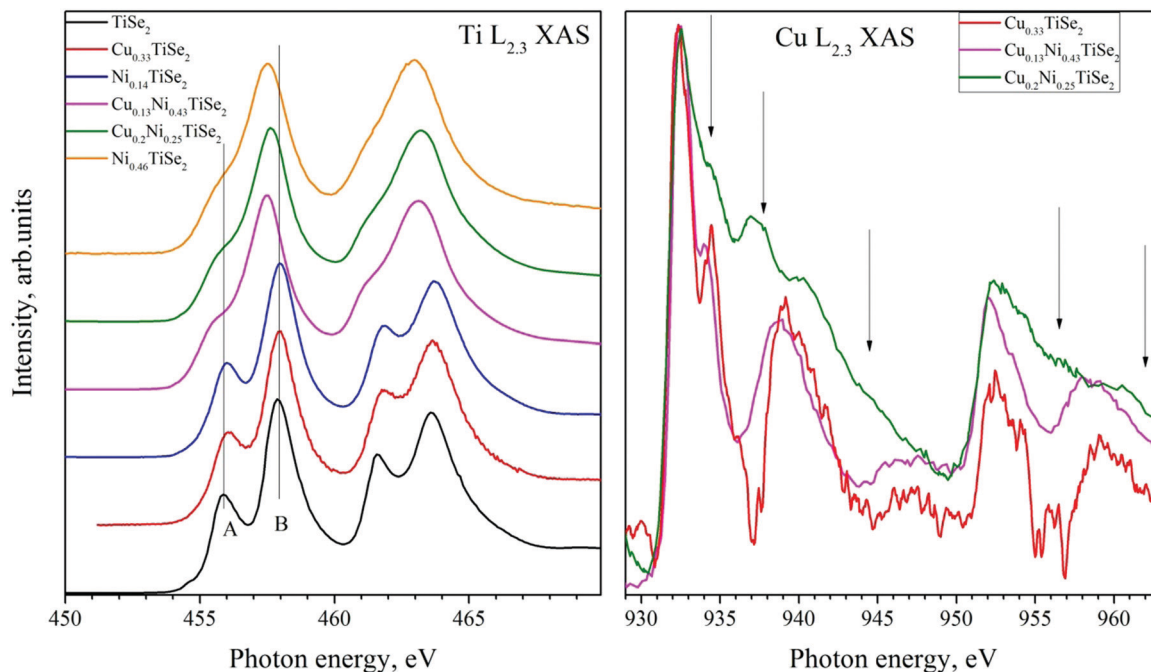


Fig. 9 Ti $L_{2,3}$ XAS for $Cu_xNi_yTiSe_2$ (left panel). Cu $L_{2,3}$ XAS for $Cu_xNi_yTiSe_2$ (right panel).

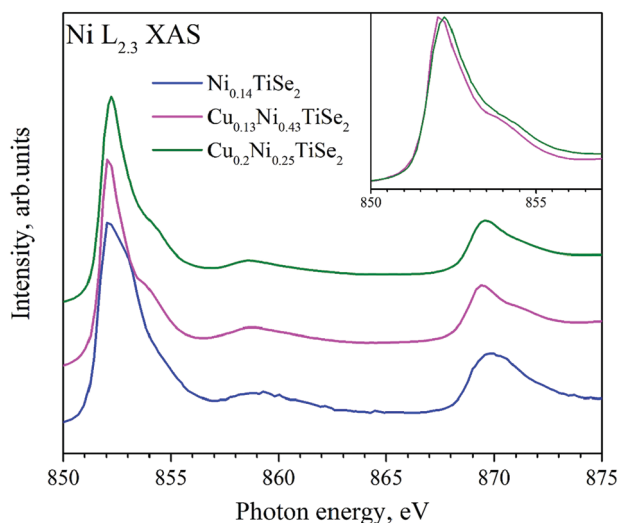


Fig. 10 Ni $L_{2,3}$ XAS for $Cu_xNi_yTiSe_2$. The inset shows Ni L_3 XAS for $Cu_{0.13}Ni_{0.43}TiSe_2$ and $Cu_{0.2}Ni_{0.25}TiSe_2$.

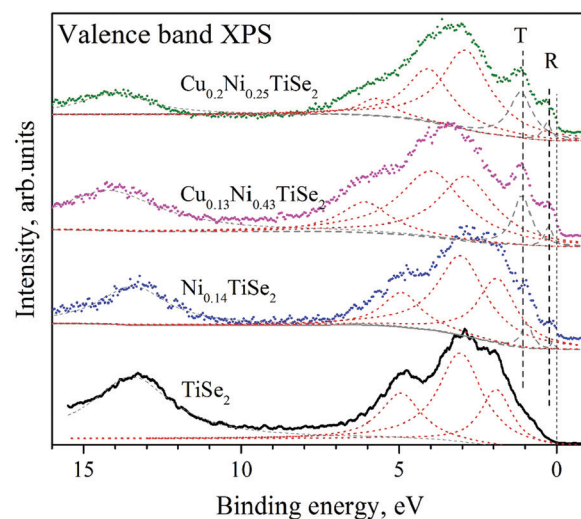


Fig. 11 The valence band spectra for $Cu_xNi_yTiSe_2$. The excitation energies are as follows: $E_{exc} = 1115$ eV for $TiSe_2$; $E_{exc} = 848.4$ eV for $Ni_{0.14}TiSe_2$; $E_{exc} = 847.8$ eV for $Cu_{0.13}Ni_{0.43}TiSe_2$; $E_{exc} = 800$ eV for $Cu_{0.2}Ni_{0.25}TiSe_2$. The peaks typical for $TiCh_2$ are shown with red dots. The T and R peaks appearing at the Cu and Ni intercalation are shown with gray dashed line.

transfer to the conduction band formed mainly by the Ti 3d states.

To obtain additional information about the states near the Fermi level, constant initial state (CIS) spectra are useful. The CIS experiment is carried out by simultaneously varying the photon energy and the photoelectron kinetic energy, *i.e.* the measurement is performed at constant ionization (or binding) energy. Ti $2p_{3/2}$ -3d CIS obtained at $E_b = 0.15$ eV from the ResPES spectra are shown in Fig. 13.

One can see that CIS are almost identical to the lineshape of Ti $L_{2,3}$ XAS for corresponding samples. For $Ni_{0.14}TiSe_2$ both

A and B peaks are clearly visible, however for $Cu_{0.13}Ni_{0.43}TiSe_2$ and $Cu_{0.2}Ni_{0.25}TiSe_2$ only the A peak is well resolved. This allows to associate these 0.15 eV-states with Ti 3d states. Different CIS for $Cu_xNi_yTiSe_2$ and Ni_yTiSe_2 indicate that Cu intercalation changes the chemical bond of the Ti atoms. Indeed, it is known that the Ti $3d_{z^2}$ orbitals hybridize with the nearest neighbors.³⁴ Since the Ti $3d_{z^2}$ orbital is directed at the octahedral site, it hybridizes with the orbitals of an atom, which is located there. However, the intensity of the resonance



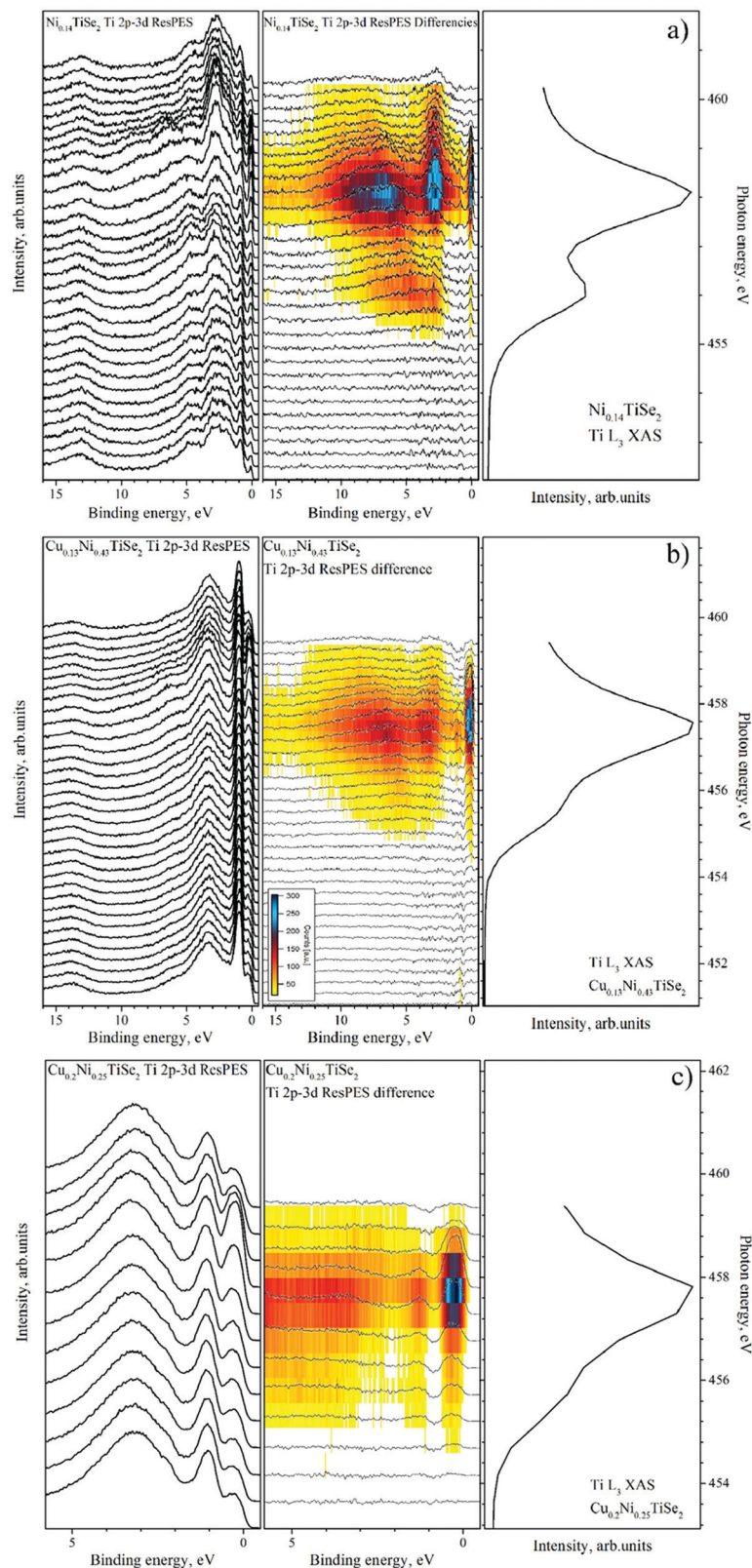


Fig. 12 The valence band spectra obtained in the Ti 2p_{3/2}–3d resonant excitation mode for Ni_{0.14}TiSe₂ (panel a), Cu_{0.13}Ni_{0.43}TiSe₂ (panel b) and Cu_{0.2}Ni_{0.25}TiSe₂ (panel c). For all the compounds (from left to right): the VB spectra obtained at different excitation energies; the difference spectra obtained by the subtraction of the off-resonance spectrum from the spectrum at the current excitation energy (as described in ref. 33); the corresponding Ti L₃ absorption spectrum.



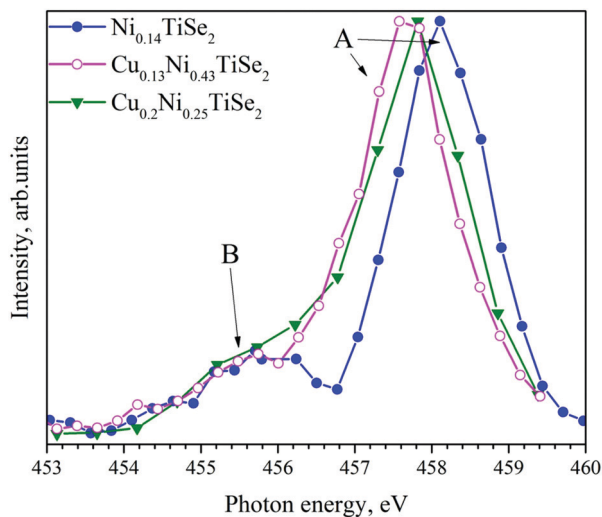


Fig. 13 Ti $2p_{3/2}$ -3d CIS at $E_b = 0.15$ eV for $Cu_xNi_yTiSe_2$ obtained from the ResPES spectra.

peak is nearly the same for all studied compounds (Fig. 14, right panel), but its width substantially increases with Cu intercalation (Fig. 14, left panel). This increase may indicate the appearance of additional hybridized states near the Fermi level.

The VB spectra obtained in the Ni $2p$ -3d resonant excitation mode are shown in Fig. 15. The main contribution is from the RRAS (resonant Raman Auger spectra) peak; however, after the subtraction of the off-resonance spectrum, a small resonance near the Fermi level both for $Cu_xNi_yTiSe_2$ and for $Ni_{0.14}TiSe_2$ can be detected at photon energies slightly beyond the resonant maximum (this peak is marked as R in Fig. 16, right panel).

Discussion

The Cu intercalation into Ni_yTiSe_2 led to unintended consequences such as the displacement of the Ni atoms from the octahedral to tetrahedral sites even at room temperature.

It would seem that this displacement of the Ni atoms to the tetrahedral sites by the Cu ones is probably caused by a Coulomb repulsion. However, the fact that the repulsion effect is observed even at $x < 0.12$ (for $y = 0.1$) and $x < 0.17$ (for $y = 0.25$), when the average distances between the intercalated atoms exceeds 18 Å, indicates, that this interaction occurs involving the host lattice.

The XPS spectra are sensitive to the effective charge of the selected atoms, however no significant changes in the Cu 2p and Ni 2p binding energies with a change in the Cu concentration are observed; the Ni and Cu charge states coincide with those in mono-intercalated compounds. At the same time, there is a sharp decrease in the Ti 2p binding energy and an increase in the Se 3d binding energy in the core-level spectra for $Cu_xNi_yTiSe_2$ as compared to those for Ni_yTiSe_2 . These changes may be caused by the redistribution of the Ni atoms over the octahedral and tetrahedral sites. As it was shown in ref. 19, the Ni 3d/Ti 3d hybridized band is formed in Ni_yTiSe_2 below the Fermi level. This band acts as a trap for the conduction electrons; therefore, the Ni doping does not increase the Fermi energy. The displacement of the Ni atoms to the tetrahedral sites makes the Ni 3d/Se 4p hybridization possible instead of the Ti 3d/Ni 3d one. To make the Ni-Se chemical bond stable, this hybridized band should lie below the Fermi level. Consequently, the displacement of the Ni atoms from the octahedral to tetrahedral sites should lead to the “return” of the Ti electrons

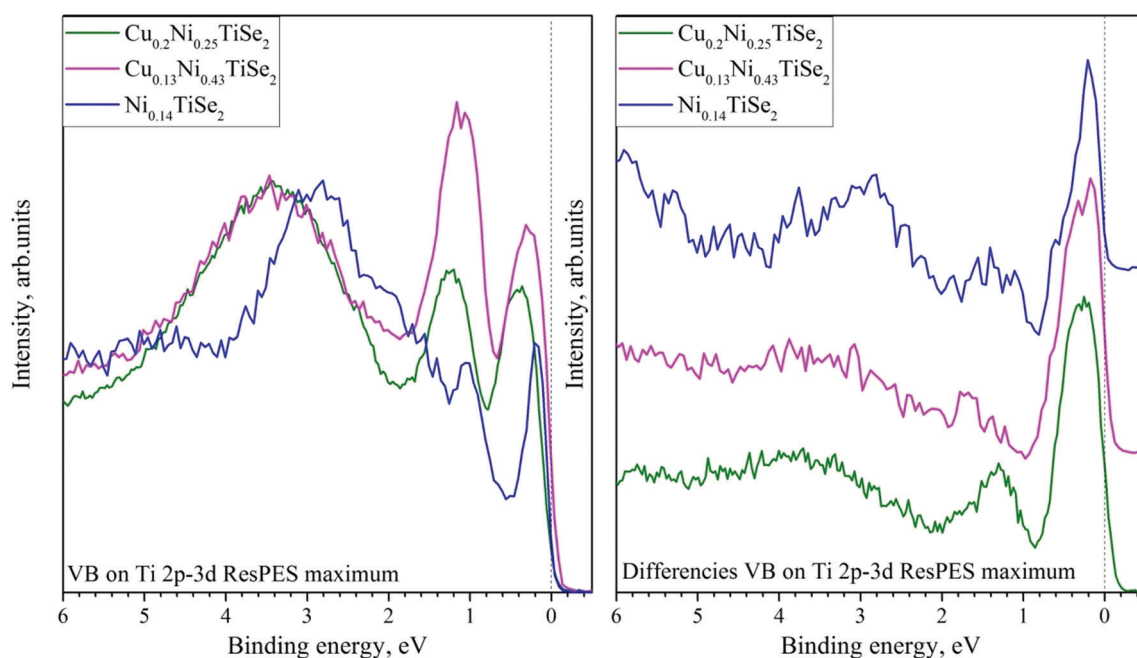


Fig. 14 The on-maximum VB spectra obtained in the Ti $2p_{3/2}$ -3d resonant excitation mode (left panel) and those after off-resonance spectra subtraction (right panel) for $Cu_xNi_yTiSe_2$. The data processing is described in detail in ref. 33.



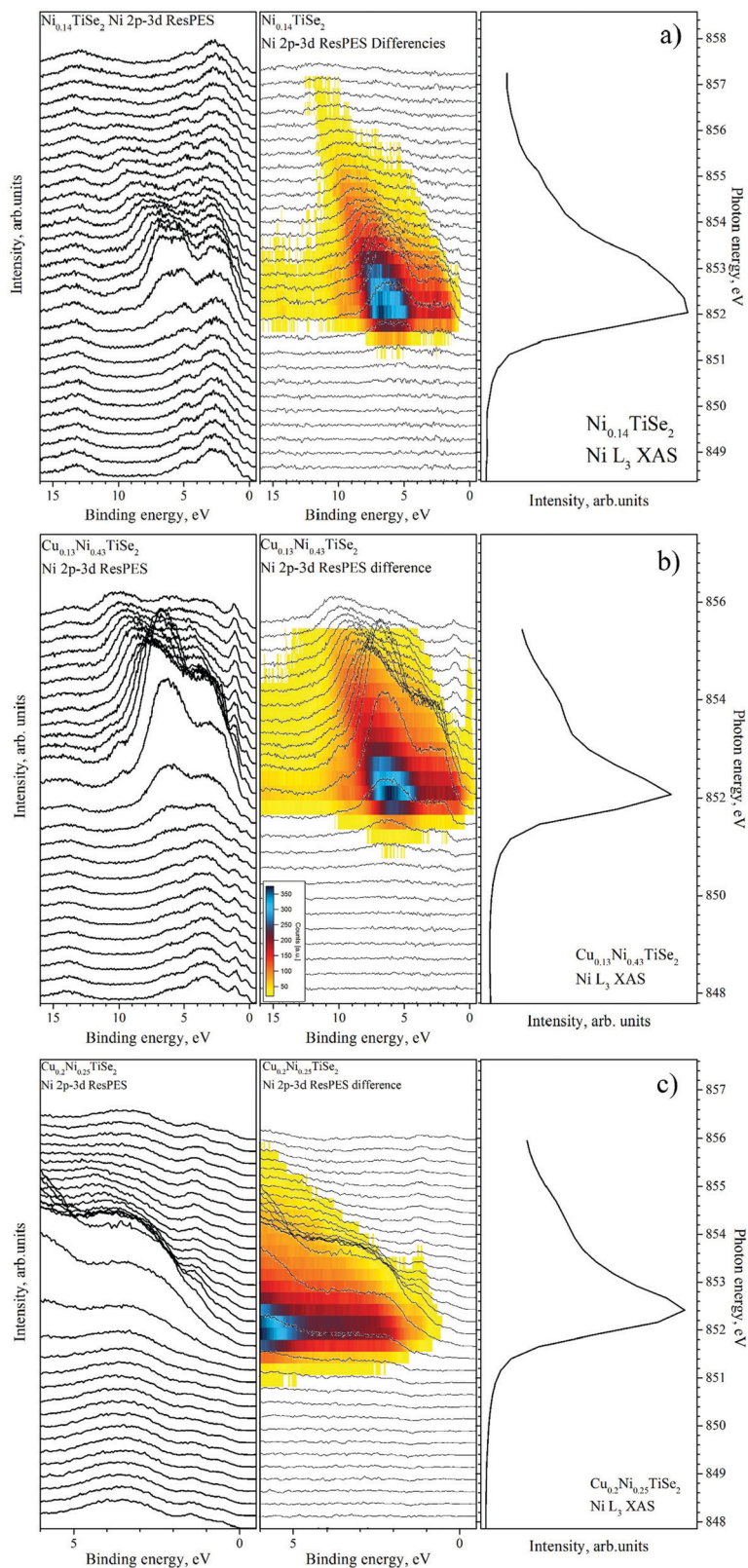


Fig. 15 The VB spectra obtained in the Ni $2p_{3/2}$ –3d resonant excitation mode for $\text{Ni}_{0.14}\text{TiSe}_2$ (panel a), $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ (panel b) and $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ (panel c). For all the compounds (from left to right): the VB spectra obtained at different excitation energies; the difference spectra obtained by the subtraction of the on-resonance spectrum from the spectrum at the current excitation energy (as described in ref. 33); the corresponding Ni L_3 absorption spectrum.



participating in the Ni 3d/Ti 3d hybridized band and trap the Se electrons participating in the Ni 3d/Se 4p hybridized band. This should decrease the Ti and Se effective charges, as indicated by the Se 3d and Ti 2p core-level spectra (Fig. 6 and 7, respectively, Table 4). This effect involves both Ni and Cu. The trend of the total (Ni + Cu) SOF for the tetrahedral sites to the constant value of approximately 1/6 indicates this (see Fig. 4, panel b).

An increase in the Se 3d binding energy should be associated with a decrease in the effective ionic charge of the selenium. Since both Ni and Cu atoms cannot be acceptors, this effect can only be associated with the formation of covalent bonds involving Ni and/or Cu states. These covalent centers could play the role of effective acceptors. As can be seen from Fig. 3 (panel b), such states are formed when Ni atoms occupy tetrahedral sites; this results in the broadening of the Se 3d line for $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$ (see Fig. 6 and Table 4).

A joint analysis of the VB spectra obtained in both resonant and non-resonant excitation modes shows that the states forming the T peak (Fig. 11) are not observed upon Ti 2p–3d and Ni 2p–3d resonance excitation, although the intensity of this peak changes noticeably. Based on the results of density of states (DOS) calculations for M_xTiCh_2 ($\text{M} = 3\text{d}$ transition metal, $\text{Ch} = \text{S}, \text{Se}, \text{Te}$),^{19,25,26,28,36} it can be assumed that the band corresponding to the T peak is formed mainly by the Se states hybridized with the Ti and Ni states. The R peak is clearly visible (Fig. 14 and 16) in the VB spectra obtained both in the Ti 2p–3d and Ni 2p–3d resonant excitation modes. This means that both the Ti 3d and Ni 3d states form the band resulting in the peak R. This is only possible when the Ni atoms occupy the octahedral sites. Therefore, the Cu intercalation leading to the displacement of the Ni atoms to the tetrahedral sites, should

decrease the intensity of the R peak. However, this displacement is accompanied by the donation of Cu electrons into the Ti 3d state. This increases the intensity of the R peak, which does not decrease, as it should be upon the displacement of the Ni atoms to the tetrahedral sites. The R peak is also observed in the $\text{Ni}_{0.14}\text{TiSe}_2$ spectra (see Fig. 11, 12, 15 and 16); however, its intensity is low because the Ti 3d states are almost empty. The Cu intercalation injects electrons to the Ti 3d states and, therefore, should increase the intensity of the R peak. We observed previously this peak and a similar effect for the Cu_xTiSe_2 system.²⁰ Thus, the increase in the intensity of the R peak caused by the filling of the Ti 3d states compensates the decrease in the intensity of the R peak caused by the decreased occupation of the octahedral sites by the Ni atoms.

As follows from the analysis of the XPS and XAS spectra for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$, the intercalated Cu atoms donate electrons to the Ti 3d-derived conduction band. This leads to an increase in the Fermi energy; as a result, the Ni 3d/Se 4p hybridized band appears below the Fermi level. This is clearly seen comparing the ResPES spectra for $\text{Ni}_{0.14}\text{TiSe}_2$ and $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$, in which Cu and Ni atoms occupy the same octahedral sites. However, an increase in the Cu concentration up to $x = 0.2$ in $\text{Cu}_{0.2}\text{Ni}_{0.13}\text{TiSe}_2$ leads both to a displacement of the Ni atoms to the tetrahedral sites and to the disappearance of the filled Ti 3d states directly at the Fermi level. Obviously, this depletion of the conduction band is associated with the absorption of electrons by a new hybridized band. Since in the observed Ni^{2+} charge state only Ni 3d_{xy} and Ni 3d_{yz} orbitals directed to the Se atoms remain unfilled, the filling of the Ni 3d shell due to the Cu intercalation leads to the amplification of the interaction between the Ni and Se atoms. Probably, it is the

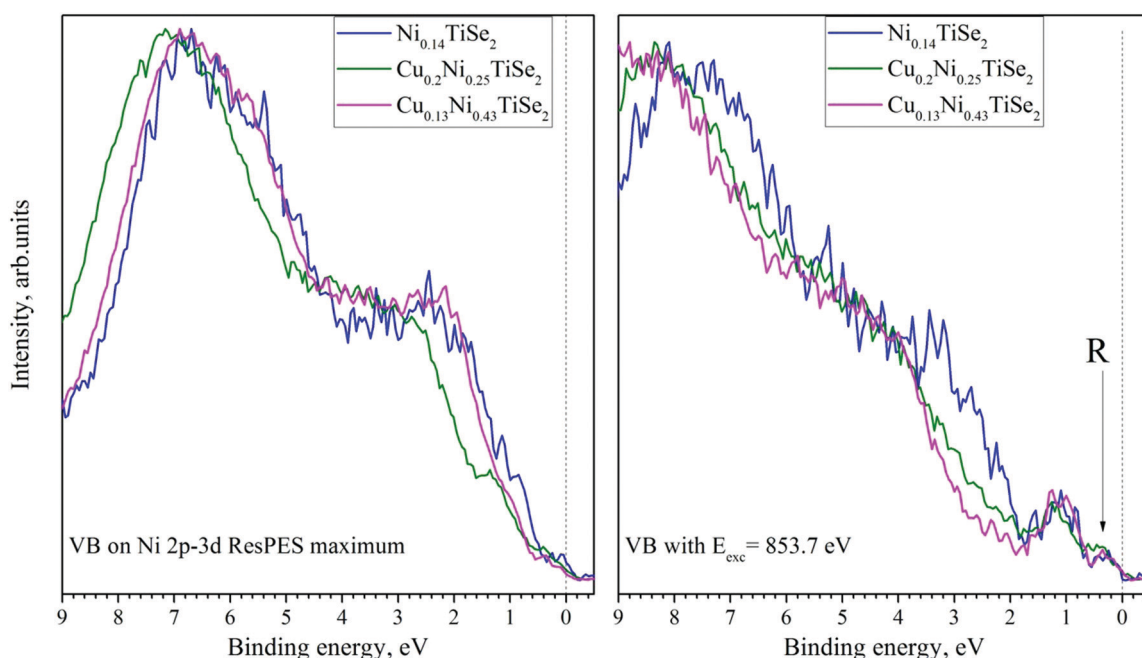


Fig. 16 The on-resonance (left panel) and near-resonance ($E_{\text{exc}} = 853.7$ eV) (right panel) VB spectra obtained in the Ni 2p_{3/2}–3d resonant excitation mode for $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ after additional processing (as described in ref. 33 and 35).



Ni 3d/Se 3p states that capture the electrons from the Ti 3d-derived conduction band.

It should be noted that in $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ the Cu atoms occupy both tetrahedral and octahedral sites, whereas in mono-intercalated Cu_xTiSe_2 all of the Cu atoms occupy only the octahedral sites. Unfortunately, we have not succeeded to prepare $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ single-crystal samples with tetrahedrally coordinated copper. In $\text{Cu}_{0.13}\text{Ni}_{0.43}\text{TiSe}_2$ the Cu atoms occupy the octahedral sites only; in $\text{Cu}_{0.2}\text{Ni}_{0.25}\text{TiSe}_2$, the Cu atoms occupy also the tetrahedral sites, but SOF for these sites is much smaller than that for the octahedral sites. We can assume two possible reasons for the redistribution of the Cu atoms in $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ over the tetrahedral and octahedral sites. The first one is a covalent interaction between the Cu and Ni atoms involving the host TiSe_2 lattice. This is supported by the trend of the total SOF for the tetrahedral sites to a “specific” value of 1/6 (Fig. 4, panel b). The second one is the perturbation of the Se sublattice by the Ni atoms, which makes the Cu 4p,3d/Se 4p hybridization in $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ more energetically favorable than that in Cu_xTiSe_2 .

It should be noted that the redistribution of the Ni atoms over the octahedral and tetrahedral sites provides only small changes in the lattice parameters upon the Cu intercalation. This confirms the previously reported assumption³⁷ that the lattice deformation upon intercalation is associated only with the occupation of the octahedral sites. The observed invariance of the lattice parameters (Fig. 2) can be useful for the application in electrochemical devices.

Conclusions

In this study, the systematic analysis of the crystal and electronic structure of the co-intercalated $\text{Cu}_x\text{Ni}_y\text{TiSe}_2$ system has been performed for the first time. It was found that the Cu intercalation into Ni_yTiSe_2 leads to the displacement of the Ni atoms from the octahedral to tetrahedral sites. This displacement can be explained by a change in the electronic structure. Being in an octahedral environment Ni^{2+} has unfilled d_{xz} and d_{yz} orbitals oriented to the chalcogen atoms. Being in a tetrahedral environment, Ni has 3 orbitals directed to the chalcogen atoms and 2 orbitals parallel to the basal plane. Obviously, the Ni atoms should balance between the occupation of octa- and tetrahedral sites; this means, respectively, the balance between the Ni 3d/Ti 3d₂/Se 3p (one Ni orbital is hybridized with Ti orbitals and two of them are hybridized with Se orbitals) and Ni 3d/Se 3p (three Ni orbitals are hybridized with Se orbitals) hybridization. An increase in the Ti 3d band filling makes the Ni 3d/Se 3p hybridization preferable, probably due to an increase in the Fermi energy and decrease in the energy gain at the Ni 3d/Ti 3d hybridization. Thus, the prediction of the preferred coordination of the intercalated atoms in these compounds should consider the spatial electronic configuration of the d-orbitals of these atoms.

The displacement of the Ni atoms from the octahedral to tetrahedral sites should also be observed in the case of the intercalation of Ni_yTiSe_2 by other donor dopants weakly perturbing the crystal structure of the host lattice. For example,

alkali metals fit well for this purpose. As a particular advantage, it can be noted that the displacement of the Ni atoms occurs at room temperature. Therefore, we expect this can be easily obtained by the electrochemical intercalation.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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