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# Unexpected high selectivity for acetate formation from CO<sub>2</sub> reduction with copper based 2D hybrid catalysts at ultralow potentials†

Rongming Cai, ab Mingzi Sun, ob Jiazheng Ren, Min Ju, ob Xia Long, ob \*a Bolong Huang \*c and Shihe Yang ob \*ab

Copper-based catalysts are efficient for  $CO_2$  reduction affording commodity chemicals. However, Cu(l) active species are easily reduced to Cu(0) during the  $CO_2RR$ , leading to a rapid decay of catalytic performance. Herein, we report a hybrid-catalyst that firmly anchors 2D-Cu metallic dots on F-doped  $Cu_xO$  nanoplates ( $Cu_xOF$ ), synthesized by electrochemical-transformation under the same conditions as the targeted  $CO_2RR$ . The as-prepared  $Cu/Cu_xOF$  hybrid showed unusual catalytic activity towards the  $CO_2RR$  for  $CH_3COO^-$  generation, with a high FE of 27% at extremely low potentials. The combined experimental and theoretical results show that nanoscale hybridization engenders an effective s,p-d coupling in  $Cu/Cu_xOF$ , raising the d-band center of Cu and thus enhancing electroactivity and selectivity for the acetate formation. This work highlights the use of electronic interactions to bias a hybrid catalyst towards a particular pathway, which is critical for tuning the activity and selectivity of copper-based catalysts for the  $CO_2RR$ .

#### Introduction

The electrocatalytic reduction of CO<sub>2</sub> (CO<sub>2</sub>RR) with low-cost catalysts holds great promise as a viable CO2 fixation process.1,2 Among the catalysts that have been extensively investigated so far, copper is unique in that it is the only metal with a negative adsorption energy for \*CO but a positive adsorption energy for \*H, favoring the formation of CO2RR products beyond CO\*.3,4 In recent years, various Cu-based electrocatalysts have been developed including metallic Cu and Cu alloys, as well as Cu compounds,5,6 and their catalytic performance has been continuously improved by surface faceting, nanostructuring, doping, etc.7-15 It is now widely accepted that the oxidation states of Cu have significant effects on the products of the CO2RR. For instance, metallic Cu is found to produce CO and HCOOH as the main products at low overpotentials but CH<sub>4</sub> or C<sub>2</sub>H<sub>4</sub> at higher overpotentials. For Cu<sub>2</sub>O, on the other hand, CH<sub>3</sub>OH is the dominant product, <sup>16</sup> and the

catalytic activity of  $\text{Cu}_2\text{O}$  decreases quickly due to the decomposition of  $\text{Cu}_2\text{O}$  to Cu.

Several approaches have been reported to stabilize Cu(1) species, such as introduction of the copper nitride (Cu<sub>3</sub>N) support, 17,18 electro-redeposition of catalysts, 19,20 and doping. 7 Moreover, the electronic and the crystalline structures of Cu compounds with multiple-anions could be greatly modulated by different charges, ionic radii and electronegativities of anions,21 leading to different catalytic performances. In fact, it has been shown theoretically and experimentally that the modulation of surface Cu(I) active sites on copper based catalysts with non-metal elements could improve the selectivity of C<sub>2</sub> products by changing the reaction pathways.<sup>8,9,11,22</sup> More recently, hybrid catalysts of Cu(0) and Cu(1) synthesized by electrochemical treatment of physically mixed Cu and CuI powder were found to enhance CO<sub>2</sub> reduction and C-C coupling to generate alcohols.10,23 However, important information on active sites and effects of anions on catalytic activity and selectivity of hybrid catalysts is still largely missing, hindering the design of efficient CO<sub>2</sub>RR catalysts by tailoring their surface/ interface properties.

Herein, we report the synthesis and the unusual  $CO_2RR$  activity/selectivity of a novel two-dimensional (2D) copper-based hybrid catalyst featuring distinctive  $F^-$  anion coordination. First, we systematically investigated the chemical and structural transformation of the 2D Cu(OH)F precursor into the hybrid catalyst under the same conditions as those for the alkaline  $CO_2RR$  (Fig. 1A). The introduction of F not only afforded the exposed high energy facets of 2D Cu(0) that were well dispersed

<sup>&</sup>quot;Guangdong Provincial Key Lab of Nano-Micro Material Research, School of Chemical Biology and Biotechnology, Shenzhen Graduate School, Peking University, Shenzhen 518055, China. E-mail: xialong@pku.edu.cn; chsyang@pku.edu.cn

<sup>&</sup>lt;sup>b</sup>Institute of Biomedical Engineering, Shenzhen Bay Laboratory, Shenzhen 518107, China

Department of Applied Biology and Chemical Technology, The Hong Kong Polytechnic University, Hung Hom, Kowloon, Hong Kong SAR, China. E-mail: bhuang@polyu.edu. hk

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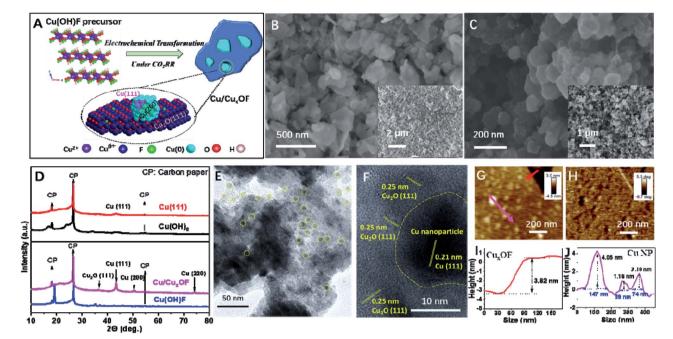


Fig. 1 Morphology and structural characterization of the as-prepared copper hybrid catalyst. (A) Schematic atomic structures of Cu(OH)F and the transformed Cu/Cu<sub>x</sub>OF hybrid. (B and C) SEM images of (B) Cu(OH)F precursor and (C) as-formed Cu/Cu<sub>x</sub>OF. (D) XRD patterns of Cu(OH)F and Cu(OH)<sub>2</sub>, and the corresponding transformed copper catalysts of Cu/Cu<sub>x</sub>OF and Cu-(111). (E) TEM image and (F) high resolution TEM image of Cu/Cu<sub>x</sub>OF. (G–J) AFM height image (G) and the corresponding height profiles of Cu<sub>x</sub>OF nanoplates (I) and metallic Cu nanoparticles (J), as well as the AFM phase image (H).

on 2D  $\rm Cu_xOF$  (named  $\rm Cu/Cu_xOF$ ), but also played a critical role in protecting  $\rm Cu(i)$  from being fully reduced to  $\rm Cu(0)$  under the reductive  $\rm CO_2RR$  conditions. Secondly, the highly active sites allowed the as-formed  $\rm Cu/Cu_xOF$  to catalyze the  $\rm CH_3COO^-$  formation via the  $\rm CO_2RR$  at an extremely low potential of -0.3 V (vs. RHE), outperforming most of the electrocatalysts reported to date for this reaction.  $^{24-27}$  In combination with theoretical calculations, we revealed the importance of the electronic interaction in terms of the s,p-d coupling between the Cu species and the hetero-anions in promoting the activity/ selectivity towards the acetate formation via the  $\rm CO_2RR$ .

#### Results and discussion

The Cu(OH)F precursor was synthesized *via* a modified hydrothermal method reported previously, <sup>28</sup> and turned out to be a well crystallized 2D compound according to the XRD patterns (Fig. S1†) and the SEM (Fig. S2† and 1B), TEM and HRTEM (Fig. S3†) images. More interestingly, the 2D microstructure was well-retained after the formation of Cu/Cu<sub>x</sub>OF *via* electrochemical transformation, although the surface showed signs of roughening (Fig. 1C). Fig. S4† shows the elemental mapping image from energy dispersive X-ray analysis (EDX), and a uniform distribution of Cu, O and F can be observed. From the X-ray diffraction (XRD) pattern of Cu/Cu<sub>x</sub>OF (Fig. 1D), typical diffraction peaks of both Cu<sub>2</sub>O and metallic Cu are observed. Apart from the strong diffraction peaks of Cu (111), other smaller peaks corresponding to (200) and (220) were also observed. Importantly, on the larger 2D nanoplates with lateral

sizes of several hundreds of nanometers, there lay much smaller ones with sizes of 20-50 nm as found in the transmission electron microscopy (TEM, Fig. 1E, S5 and Table S1†), high resolution TEM (HRTEM, Fig. 1F) and atomic force microscopy (AFM, Fig. 1G-J and S6†) images. Such an overall morphology of Cu/Cu<sub>x</sub>OF is illustrated in Fig. 1A. The lattice fringes of  $\sim 0.25$  nm,  $\sim 0.18$  nm and 0.21 nm in the HRTEM images (Fig. 1F and S7†) further suggest that the larger nanoplates are made of Cu<sub>2</sub>O with the lattice fringe of (111), and the smaller 2D nanoparticles are actually metallic Cu with the lattice fringes of (200) and (111), respectively, in accordance with the XRD results. Note that the as-formed Cu<sub>x</sub>OF nanoplates comprise both well crystallized Cu2O grains and amorphous regions, which could be well distinguished from each other in the HRTEM images (Fig. 1F and S8†). This is also consistent with the low peak intensity of Cu<sub>2</sub>O in the XRD patterns (Fig. 1D).

In a control experiment, we also investigated the structural change of Cu(OH)F in 1 M KOH electrolyte, as opposed to the abovementioned electrochemical transformation. From the XRD patterns shown in Fig. S9,† one can see that after the KOH treatment, Cu(OH)F was transformed into  $Cu(OH)_2$  with a typical nanowire microstructure (Fig. S10 and S11†). This is supported by the greatly reduced content of F ( $\sim$ 0.24%, see Fig. S12†). Of note, the much higher F content in  $Cu/Cu_xOF$  is probably due to the fast transformation process as confirmed by *in situ* electrochemical quartz crystal microbalance with dissipation (EQCM-D) (Fig. S13†). These results indicate that the direct electrochemical treatment was the main driving force for the successful formation of  $Cu/Cu_xOF$  (with the F content as

high as  ${\sim}5.92\%$ ) from the Cu(OH)F precursor (Fig. S14†). Further, Cu(OH)<sub>2</sub> without any F (Fig. S15†) and KOH-treated Cu(OH)F (with  ${\sim}0.24\%$  F) were subsequently treated under the CO<sub>2</sub>RR conditions. As can be seen from the XRD patterns (Fig. 1D and S16†), HRTEM (Fig. S17 and S18†) and AFM (Fig. S19†) images, only 2D metallic Cu(0) with the low-energy facet of (111) could be obtained (named Cu-(111)) from the Cu(OH)<sub>2</sub> precursor, while Cu/Cu<sub>x</sub>O could be formed by using the KOH-treated Cu(OH)F as the precursor, further confirming the critical role of F in stabilizing the oxidized Cu species.

Then X-ray photoelectron spectroscopy (XPS), Auger electron spectroscopy (AES), and electron paramagnetic resonance (EPR) spectroscopy were combined to get full information on the electronic structure of the metal and anions in the as-prepared copper catalysts. First, the XPS peak of F in Cu/Cu<sub>r</sub>OF was much sharper than that of the Cu(OH)F precursor (Fig. 2A), suggesting a different electronic environment of F in Cu/Cu<sub>r</sub>OF arising from the decreased content of F and the absence of H<sup>+</sup> ions. The binding energy of O 1s in Cu/Cu<sub>x</sub>OF was slightly larger than that in Cu(OH)<sub>2</sub> while smaller than that in Cu(OH)F (Fig. 2B), signifying the electron withdrawing and polarization effects of F on O 1s, and expectedly, on the directly bonded metal ions, i.e.,  $Cu^{\delta^+,29}$  Moreover, negligible peaks could be found for the O spin-orbital of Cu-(111) (Fig. S20†), indicating the existence of only metallic Cu without any oxidized copper species, in accordance with the results of XRD (Fig. 1D), HRTEM (Fig. S17†) and XPS of Cu-(111) in the Cu 2p spin-orbital (Fig. 2C). As expected, for the Cu(OH)<sub>2</sub> precursor, Cu<sup>2+</sup> located at 935.5 eV was clearly observed. Notably, a large positive shift of Cu 2p was observed for Cu(OH)F (Fig. S21†) due to the strong polarization and electron withdrawing effects of F as mentioned above. For the as-prepared Cu/Cu<sub>x</sub>OF, however, we found that besides Cu(0), there also existed moieties between Cu(0) and Cu(1), hereafter denoted as Cu $^{\delta+}$  (to distinguish them from Cu and Cu(1) in Cu/Cu<sub>2</sub>O), as shown in Fig. 2C, D and S22.† The LMM Auger spectra of copper shown in Fig. 2E further confirm the existence of metallic Cu(0) and Cu $^{\delta+}$  in the as-prepared Cu/Cu<sub>x</sub>OF. These results suggest that in addition to the electronic interactions between O and F, the two anions together also impact the Cu species, and thus would influence the catalytic performance of the hybrid catalyst.

Further, EPR spectra were collected and analyzed in order to get detailed information on the unpaired electrons in the material. The *g*-factor is a dimensionless measure of the intrinsic magnetic moments of the electron, which is 2.0023 for a free electron but can vary for unpaired electrons in different molecules and materials. Here Cu(1) has a d<sup>10</sup> electronic configuration with no unpaired electrons and thus is EPR silent. However, EPR is a useful tool for both structural and mechanistic studies of Cu(11) and Cu(0) on account of their d<sup>9</sup> and d<sup>10</sup>s¹ configurations, respectively. As can be seen from Fig. S23,† Cu(111) and Cu/Cu<sub>x</sub>OF show a *g* value close to ~2.63, indicating the presence of unpaired electrons in both the materials of similar origin, namely the metallic Cu species.

We examined the stability of the catalysts by checking any possible structural changes. As shown in Fig. 2F, all the XPS peaks for F, O, and Cu spin-orbitals of Cu/Cu<sub>x</sub>OF showed a negligible change after the electrochemical CO<sub>2</sub>RR evaluation

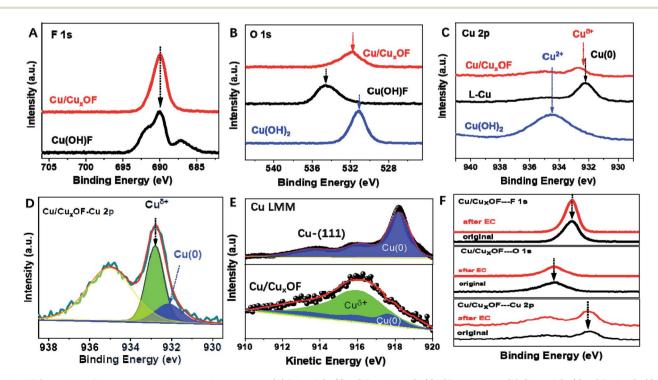


Fig. 2 XPS analysis of the copper catalysts and precursors. (A) F 1s of  $Cu/Cu_xOF$  and the Cu(OH)F precursor, (B) O 1s of  $Cu/Cu_xOF$ , the Cu(OH)F precursor and  $Cu(OH)_2$  precursor, (C)  $Cu_xOF$ ,  $Cu_xOF$ ,  $Cu_xOF$ ,  $Cu_xOF$ ,  $Cu_xOF$ , (D) de-convoluted XPS spectrum of  $Cu_xOF$  and  $Cu_xOF$ , (E)  $Cu_xOF$ , (E)  $Cu_xOF$  at F 1s, O 1s, and  $Cu_xOF$  and  $Cu_xOF$  and  $Cu_xOF$  at F 1s, O 1s, and  $Cu_xOF$  are spin-orbitals before and after the electrochemical tests at the stepped potentials from  $Cu_xOF$  and  $Cu_xOF$  are  $Cu_xOF$  and  $Cu_xOF$  are  $Cu_xOF$  and  $Cu_xOF$  and  $Cu_xOF$  are  $Cu_xOF$  and  $Cu_xOF$  and  $Cu_xOF$  are  $Cu_xOF$  are  $Cu_xOF$  and  $Cu_xOF$  are  $Cu_xOF$  an

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at the stepped potentials from -0.3 V to -0.7 V vs. RHE for more than 28 h. Moreover, based on the XRD (Fig. S24†), TEM (Fig. S25†), HRTEM (Fig. S26†) and elemental analysis (Fig. S27 & S28†) results, we found that both the metallic Cu and 2D Cu<sub>x</sub>OF with  $\sim 5.83\%$  of F (close to the initial value of  $\sim 5.92\%$ ) were retained after the  $\sim 28$  h stability tests, indicating the good working stability of the as-prepared Cu/Cu<sub>x</sub>OF for the CO<sub>2</sub>RR. In addition, the *in situ* EQCM-D further suggests that the surface of Cu/Cu<sub>x</sub>OF is actually in a dynamic equilibrium as characterized by periodic adsorption–desorption processes during the CV cycles (Fig. S29†).<sup>30,31</sup>

Then we proceeded to the electrochemical performance of the as-prepared copper catalysts towards the CO<sub>2</sub>RR, which was evaluated in a flow cell with 1 M KOH as the electrolyte (see the ESI† for details). From the *I–V* curves (Fig. 3A), it is clear that Cu/ Cu<sub>x</sub>OF showed larger current densities than Cu-(111) and Cu/ Cu<sub>r</sub>O at the same applied potentials. Moreover, the partial current density (Fig. S30†) and faradaic efficiency (FE, Fig. 3B) of  $CH_3COO^-$  on  $Cu/Cu_xOF$  were found to be  $\sim 4.0$  mA cm<sup>-2</sup> and 27% at a low potential of -0.3 V (vs. RHE), which differs from those of the main C2 products of C2H5OH and C2H4 on Cu-(111) (Fig. S31†) and Cu/Cu<sub>x</sub>O (Fig. S32†), and also competes with the best noble-metal free electrocatalysts for the CO<sub>2</sub>RR with the selective product of CH3COO that usually required more negative potentials (Table S2†). 16,24-27,32 We noted that Cu/Cu<sub>r</sub>OF and Cu/Cu<sub>x</sub>O have a similar microstructure including both nanoparticle size and hierarchical structure (Fig. S33-S35†), thus the distinctive catalytic selectivity of Cu/Cu<sub>r</sub>OF for the CO<sub>2</sub>RR (Fig. 3E-I) probably results from the F-doping induced modification of surface physicochemical properties of Cu/  $Cu_xOF$ , which would favor the acetate formation. Moreover, the current density of the CO<sub>2</sub>RR reached  $\sim$ 56 mA cm<sup>-2</sup> at E =-0.5 V (vs. RHE) (Fig. 3A), larger than the reported values

collected on copper based catalysts under similar reaction conditions. 13,33,34 In addition, compared with Cu-(111) and Cu/ CuxO, Cu/CuxOF showed much higher FE for the CO2RR (Fig. 3C) and lower FE for the HER (Fig. 3D) in the whole potential range from -0.3 to -0.7 V vs. RHE, demonstrating much better CO<sub>2</sub>RR activity of Cu/Cu<sub>r</sub>OF than Cu-(111) and Cu/ Cu<sub>x</sub>O catalysts. The long-term stability at stepped potentials was also estimated. From Fig. S36,† it is clear that the change of the CO2RR current density at each stage is less than 5% even at a deep potential down to  $-0.7 \text{ V} \nu s$ . RHE. What's more, the FE of acetate during the stability tests was essentially unchanged at each potential, evidencing the good catalytic stability of the asprepared Cu/Cu<sub>x</sub>OF for the CO<sub>2</sub>RR. In conjunction with the structural stability discussed earlier in the previous paragraph, we can conclude that the as-prepared Cu/Cu<sub>r</sub>OF is indeed an advanced electrocatalyst for the CO2RR with excellent catalytic selectivity and stability toward acetate generation.

As is widely known, the CO<sub>2</sub>RR involves multiple electrontransfer and chemical reaction steps, and this complex process strongly depends on the physiochemical and surface properties of the catalysts. Previous theoretical and experimental studies have demonstrated that the products of the CO<sub>2</sub>RR at low overpotentials were generally limited to C<sub>1</sub> species on metallic Cu nanocatalysts, especially on their low-energy facets, such as CO and HCOOH. 35,36 This was in accordance with the results observed for Cu-(111) prepared in this work. At high potentials, the FE of C2 increased along with the suppression of the HER, which again agrees with what we observed for Cu-(111) (Fig. 3G, I and S37†).37,38 However, markedly different catalytic selectivities were observed for Cu/  $Cu_xOF$ . Specifically, the FE of  $C_2$  products was maintained at  $\sim$ 30% in almost the whole potential range we studied, while the FE of C<sub>1</sub> products increased (Fig. 3G and F) in parallel with the

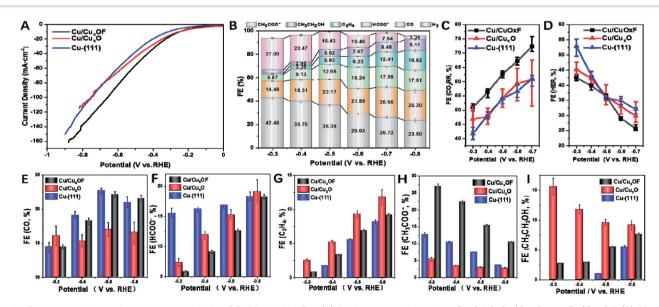


Fig. 3 Electrocatalytic performance towards the CO<sub>2</sub>RR in 1 M KOH. (A) J-V curves collected on Cu-(111), Cu/Cu<sub>x</sub>O and Cu/Cu<sub>x</sub>OF. (B) FE of all the products on Cu/Cu<sub>x</sub>OF at various applied potentials. (C and D) FE for (C) the CO<sub>2</sub>RR and (D) HER on Cu-(111), Cu/Cu<sub>x</sub>O and Cu/Cu<sub>x</sub>OF. (E–I) comparison of FE on Cu-(111), Cu/Cu<sub>x</sub>O and Cu/Cu<sub>x</sub>OF for all the CO<sub>2</sub>RR products of (E) CO, (F) HCOO<sup>-</sup>, (G) C<sub>2</sub>H<sub>4</sub>, (H) CH<sub>3</sub>COO<sup>-</sup> and (I) CH<sub>3</sub>CH<sub>2</sub>OH.

suppression of the HER when the applied potential increased (Fig. 3E–I), amounting to  $\sim$ 72% FE at -0.7 V  $\nu s$ . RHE (Fig. 3B and C). So what is the explanation for this striking catalytic performance? Plausibly, in Cu/Cu<sub>x</sub>OF with F-coordination, multi-copper oxidation states and the hierarchical 2D–2D microstructure, could provide multiple sites with different adsorption–desorption characteristics for the reactants and intermediates of the CO<sub>2</sub>RR (Scheme S1†), promoting C–C coupling<sup>9,39–41</sup> to form the CH<sub>3</sub>COO $^-$  product even at very low overpotentials. More detailed discussions on the reaction mechanism along with theoretical calculations will be provided below.

To better understand the  $\mathrm{CO_2RR}$  process on the copperbased hybrid catalysts, we carried out the DFT calculations of their electronic structures and energy trends in the reaction process of interest here, and then explored the effects of the electronic structure and energy on electroactivity of the catalysts (Fig. S38†). Clearly, the Cu-(111) surface shows a highly ordered electronic distribution guided by the Cu sites (Fig. 4A). Such a surface electronic structure usually leads to low product selectivity due to the strong competition between different reaction pathways. By contrast, the  $\mathrm{Cu/Cu_xOF}$  surface shows a strong perturbation of the electronic distribution near the Fermi level ( $E_F$ ) (Fig. 4B) due to hybridization with the  $Cu_xOF$  nanoplate. We believe that the resulting strong bonding orbitals of the metallic Cu nanoparticle surfaces played an important role in forming the highly electroactive region for the  $CO_2RR$ . To illustrate this point more clearly on  $Cu/Cu_xOF$ , Fig. 4C displays the projected partial density of states (PDOS). Notably, the Cu-3d orbitals in  $Cu/Cu_xOF$  are now located at a position much closer to the  $E_F$  than those in Cu-(111), meaning a higher d-band center and thus an improved electroactivity. Meanwhile, both the O-s,p, and the F-s,p orbitals are located deeper in energy, which acted as the electron reservoir. Compared to  $Cu/Cu_xOF$ ,  $Cu/Cu_xO$  shows a slightly lower d-band center, which leads to decreased electroactivity and FE of the  $C_2$  products (Fig. S39†).

We next unravel the site-dependent PDOS of the different elements in the catalysts. Focusing now on the Cu sites, we notice that from the bulk  $Cu_xOF$  nanoplate to the surface of metallic Cu with high-energy facets, Cu-3d orbitals exhibited a nearly linear upshifting trend toward the  $E_F$  (Fig. 4D). Consequently, the surface Cu nanoparticles with high electroactivity were found to be the active sites for the  $CO_2RR$  process. Then, the electronic structures of both O and F sites in the  $Cu_xOF$ 

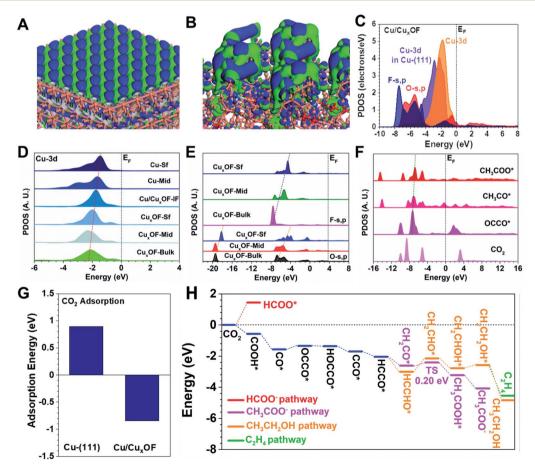


Fig. 4 The 3D contour plot for electronic distributions of (A) Cu-(111) and (B) Cu/Cu<sub>x</sub>OF. (C) PDOS of Cu/Cu<sub>x</sub>OF. Site-dependent PDOS of (D) Cu-3d, and (E) O-s,p and F-s,p in Cu/Cu<sub>x</sub>OF. (F) The PDOS of key adsorbates during the CO<sub>2</sub>RR process. (G) The comparison of CO<sub>2</sub> adsorption on Cu-(111) and Cu/Cu<sub>x</sub>OF. (H) The CO<sub>2</sub>RR reaction pathways on Cu/Cu<sub>x</sub>OF. The orange, red, and cyan balls in (A) and (B) represent Cu, O and F atoms, respectively, and the blue and green isosurfaces indicate the bonding and anti-bonding orbitals, respectively. The dashed lines in (D and E) represent the higher position of the band centers for Cu and O-s,p, and F-s,p, respectively.

nanoplates are investigated (Fig. 4E). Notably, from the bulk to the nanoplate surface, the s,p orbitals exhibit a gradual upshifting trend, which supports the improved electron transfer capability from the Cu<sub>x</sub>OF surface to the Cu nanoparticle surface, and ultimately to the intermediates during the CO<sub>2</sub>RR. To follow up the intermediate conversion process, the PDOS of key adsorbates during the CO2RR process are displayed in Fig. 4F. The structural configurations of these key adsorbates are displayed in Fig. S40.† Compared to the free CO<sub>2</sub>, the s,p orbitals of the intermediates have shown a slight upshift, confirming the successful reduction of CO<sub>2</sub>. In the reduction of OCCO\* to CH<sub>3</sub>COO\*, we notice a nearly linear correlation of the s,p orbital energies, a sign that would guarantee efficient electron transfer and the intermediate conversions. The linear correlation of the  $\boldsymbol{\sigma}$ orbitals with an upshifting trend is well preserved, which not only indicates the most efficient electron transfer but also the optimal adsorption strength of intermediates for the CO<sub>2</sub>RR.

Finally, the adsorption of CO<sub>2</sub> on the catalysts was also investigated. Compared to the Cu-(111) surface, the adsorption on Cu/Cu<sub>x</sub>OF became much more energetically favorable (Fig. 4G). While the CO<sub>2</sub> adsorption on the Cu-(111) surface showed an energy cost as high as 0.89 eV, Cu/Cu<sub>x</sub>OF demonstrates a highly exothermic adsorption energy of -0.84 eV, accounting for the high electroactivity towards the CO<sub>2</sub>RR. The energy evolution of the CO<sub>2</sub>RR process is shown in Fig. 4H. For the initial hydrogenation, the strong preference of O-H over C-H shows a low selectivity towards the formation of HCOO on Cu/ Cu<sub>r</sub>OF. The key reaction step in formation of C<sub>2</sub> products usually relies on the coupling of the CO\* intermediates to form OCCO\*, which shows only a minor energy barrier of 0.22 eV. The following reduction steps are all energetically favorable until the formation of HCCO\*. In further hydrogenation, the formation of both CH<sub>2</sub>CO\* and HCCHO\* becomes exothermic. The key step in acetic acid formation is the incorporation of water in CH<sub>2</sub>CO\*, which is energetically favorable with a small activation barrier of 0.20 eV. This low barrier guarantees the efficient conversion towards CH<sub>3</sub>COOH\*. Remarkably, the strong reaction trend with an energy release of −4.08 eV makes the formation of CH<sub>3</sub>COO the most preferred of all the reaction pathways at a very low potential, in good agreement with the foregoing experimental findings. Proceeding with the CH<sub>3</sub>CH<sub>2</sub>OH and C<sub>2</sub>H<sub>4</sub> reaction pathways, the hydrogenation steps from HCCHO\* to CH2CHO\* and from CH2CHOH\* to CH2CH2OH\* have an energy barrier of 0.87 and 0.22 eV, respectively. These substantial energy barriers, especially the first one, largely limit the formation of both CH<sub>3</sub>CH<sub>2</sub>OH and C<sub>2</sub>H<sub>4</sub> at low potentials. However, by increasing the applied potential, the contents of CH<sub>3</sub>CH<sub>2</sub>OH and C<sub>2</sub>H<sub>4</sub> will gradually increase, as we observed experimentally. Therefore, both electronic structures and reaction trends have confirmed the high electroactivity and selectivity of Cu/Cu<sub>x</sub>OF towards the generation of C<sub>2</sub> products during the CO<sub>2</sub>RR.

#### Conclusions

In summary, by exploiting 2D Cu(OH)F nanoplates as a precursor, we successfully synthesized a copper based hybrid catalyst, by coupling 2D metallic Cu nanoparticles with high-

energy facets and F-doped CuxO (CuxOF) nanoplates. For the catalyst synthesis with in situ electrochemical transformation, we purposely used the same conditions as those for the alkaline CO<sub>2</sub>RR in order to ensure a durable catalytic operation. The critical roles of F-modification have been revealed to effectively tailor the exposed facets of metallic Cu nanoparticles, stabilize the oxidized copper active species under the CO<sub>2</sub>RR conditions, and more importantly, to purposely induce the s,p-d coupling between the metal and hetero-anions tending heavily towards the acetate pathway. Consequently, the as-prepared 2D Cu/ Cu<sub>r</sub>OF hybrid catalyst creates an electronic environment leading to high electroactivity, particularly the unexpected CH<sub>3</sub>COO<sup>-</sup> selectivity at an extremely low overpotential. This work unravels interesting electronic interactions between Cu species and different anions in copper-based hybrid materials, and provides an efficient strategy to construct more efficient catalysts for the CO2RR.

# Data availability

All experimental and computational data are available within the article or in the ESI file.†

### **Author contributions**

X. L. and S. Y. conceived the idea. R. C. synthesized the catalysts. R. C. and J. R. carried out structural and electrochemical characterization. R. C. and X. L. performed EQCM-D experiments. R. C., M. J. and X. L. performed AFM characterization experiments. R. C., M. J., and J. R. performed morphology characterization. X. L. and R. C. prepared the figures and analyzed the data. M. S. and B. H. performed the DFT calculations. X. L., R. C. B. H. and S. Y. wrote the manuscript. All the authors discussed the results and commented on the manuscript at all stages.

# Conflicts of interest

There are no conflicts to declare.

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