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Influence of the primary and secondary coordination spheres on nitric oxide adsorption and reactivity in cobalt(II)–triazolate frameworks†

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Nitric oxide (NO) is an important signaling molecule in biological systems, and as such, the ability of porous materials to reversibly adsorb NO is of interest for potential medical applications. Although certain metal–organic frameworks are known to bind NO reversibly at coordinatively unsaturated metal sites, the influence of the metal coordination environment on NO adsorption has not been studied in detail. Here, we examine NO adsorption in the frameworks $\text{Co}_2\text{Cl}_2(\text{bbta})$ ($\text{H}_2\text{bbta} = 1H,5H\text{-benzo}(1,2\text{-}d':4,5\text{-}d'')\text{bistriazole}$) and $\text{Co}_2(\text{OH})_2(\text{bbta})$ using gas adsorption, infrared spectroscopy, powder X-ray diffraction, and magnetometry. At room temperature, NO adsorbs reversibly in $\text{Co}_2\text{Cl}_2(\text{bbta})$ without electron transfer, with low temperature data supporting spin-crossover of the NO-bound cobalt(II) centers of the material. In contrast, adsorption of low pressures of NO in $\text{Co}_2(\text{OH})_2(\text{bbta})$ is accompanied by charge transfer from the cobalt(II) centers to form a cobalt(III)–NO[−] adduct, as supported by diffraction and infrared spectroscopy data. At higher pressures of NO, characterization data indicate additional uptake of the gas and disproportionation of the bound NO to form a cobalt(III)–nitro (NO₂[−]) species and N₂O gas, a transformation that appears to be facilitated by secondary sphere hydrogen bonding interactions between the bound NO₂[−] and framework hydroxo groups. These results provide a rare example of reductive NO binding in a cobalt-based metal–organic framework, and they demonstrate that NO uptake can be tuned by changing the primary and secondary coordination environment of the framework metal centers.

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Introduction

Nitric oxide (NO) is recognized as a key gasotransmitter with a number of important physiological roles, including as the endothelium-derived relaxing factor and as a neurotransmitter involved in memory formation and stroke damage.^{1,2} Given its influence on vasodilation, administration of NO gas and NO donor drugs are an important therapeutic strategy, and NO gas administration is under investigation as a potential supportive treatment for coronaviruses, in particular SARS-CoV-2.^{3–6} The reactivity of NO with proteins containing metal cofactors is critical to the therapeutic effects of this gas.^{7–10} For example, metal-containing biomolecules, including guanylate cyclase,¹¹ cytochrome *c* oxidase,¹² hemoglobin,^{13,14} and cobalamin,^{15–18} bind nitric oxide at the metal site, an event that can either

enable or interfere with critical metabolic pathways. In such metalloproteins, nitric oxide binding is facilitated by the low reduction potentials of the transition metal-containing cofactors to bind NO *via* an electron-transfer mechanism, generating an oxidized metal and a reduced NO adduct.^{7,14,19–21} However, because of the deleterious reactions that can occur in the presence of excess nitric oxide, careful and controlled delivery is an active area of research.²²

Porous solid-state materials have generated considerable interest as NO-releasing therapeutics due to their tunability and high surface areas, and metal–organic frameworks in particular represent a key class of materials under investigation.^{22–24} Composed of metal ion or cluster nodes and multitopic organic linkers, metal–organic frameworks display high degrees of thermal and chemical stability in addition to considerable chemical and structural variability.^{25,26} Adsorption of NO has been reported in a number of frameworks, including $\text{Cu}_3(\text{btc})_2$ (HKUST-1; $\text{btc}^{3-} = 1,3,5\text{-benzenetricarboxylate}$),²⁷ $\text{M}_2(\text{dobdc})$ ($\text{M} = \text{Mg}, \text{Co}, \text{Ni}, \text{and Zn}$; $\text{dobdc}^{4-} = 2,5\text{-dioxido-1,4-benzenedicarboxylate}$),^{28–32} and $\text{Zr}_6\text{O}_4(\text{OH})_4(\text{bdc})_6$ (UiO-66; $\text{bdc}^{2-} = 1,4\text{-benzenedicarboxylate}$) and its derivatives.³³ Many of these materials physisorb NO at coordinatively-unsaturated metal centers, with little evidence of charge transfer. However, reduction of nitric oxide upon adsorption has been

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characterized in copper(i)- and iron(ii)-containing frameworks, such as $\text{Cu}_2\text{Zn}_3\text{Cl}_2(\text{btdd})$ ($\text{Cu}^{\text{I}}\text{-MFU-4l}$; $\text{H}_2\text{btdd} = \text{bis}(1H\text{-}1,2,3\text{-triazolo}[4,5\text{-}b], [4',5'\text{-}i])\text{dibenzo}[1,4]\text{dioxin}$),³⁴ $\text{Cu}^{\text{I}}\text{-ZrTpmC}^*$ ($\{[\text{Zr}_6\text{O}_4(\text{OH})_4(\text{C}_6\text{H}_5\text{COO})_4]_3(\text{Cu}(\text{CH}_3\text{CN})\text{TpmC}^*)_8\}$, where $\text{TpmC}^* = 1,1',1''\text{-methanetriyltris-(3,5-dimethyl-1H-pyrazole-4-carboxylic acid)}$),³⁵ $\text{Fe}_2(\text{dobdc})$,²⁸ MIL-88(Fe) derivatives ($\text{Fe}_3\text{OX(L)}$, where $\text{X} = \text{F, Cl, or OH}$ and $\text{L}^{2-} = \text{fumarate}$),³⁶ $\text{Fe}(\text{bddpi})(\text{DMF})$ ($\text{bddpi}^{2-} = 2,6\text{-bis}(1H\text{-pyrazolyl})\text{-pyromellitic diimide}$; $\text{DMF} = N,N'\text{-dimethylformamide}$),³⁷ and $[\text{Zn}_3\text{-FeO}(\text{bdc})_3]_8$ (Fe-MOF-5).^{38,39} In the case of $\text{Fe}_2(\text{dobdc})$ and MIL-88(Fe) , the NO unit is formally retained upon adsorption and is released in the presence of humidity.^{28,36} In contrast, upon binding at the copper(i) sites in Cu-MFU-4l and $\text{Cu}^{\text{I}}\text{-ZrTpmC}^*$ or the iron(ii) sites in Fe-MOF-5 , three equivalents of NO disproportionate to form N_2O and a metal-bound NO_2^- species.^{34,35,38} Characteristic of most of these examples is that NO binding is strengthened exclusively *via* an electron transfer mechanism, in the absence of contributions from favorable non-covalent interactions.

Recently, we demonstrated that the framework $\text{Co}_2(\text{OH})_2(\text{bbta})$ ($\text{H}_2\text{bbta} = 1H,5H\text{-benzo}(1,2\text{-}d:4,5\text{-}d')\text{bistriazole}$; Fig. 1c) reversibly binds O_2 *via* electron transfer to form a cobalt(III)-superoxo adduct that is stabilized by hydrogen bonding with the bridging hydroxo groups of the framework.⁴⁰ In contrast, the related framework $\text{Co}_2\text{Cl}_2(\text{bbta})$ (Fig. 1a) exhibits much weaker uptake of O_2 with no associated electron transfer. Here, we show that synergistic electron transfer and stabilizing hydrogen bonding interactions can also promote strong binding of NO in $\text{Co}_2(\text{OH})_2(\text{bbta})$ (Fig. 1d-f), and, at higher pressures, disproportionation of NO to yield a cobalt(III)-nitro (NO_2^-) species. Additionally, the framework $\text{Co}_2\text{Cl}_2(\text{bbta})$ is shown to exhibit strong and reversible adsorption of NO at 298 K (Fig. 1e and f), with evidence for some low-spin cobalt(II) character at low temperatures (100 K), due to back-bonding from bound NO. The unique binding of NO in these frameworks is characterized by gas adsorption, infrared spectroscopy, powder X-ray diffraction, and magnetometry measurements.

Fig. 1 Structures of $\text{Co}_2\text{Cl}_2(\text{bbta})$ (a) and $\text{Co}_2(\text{OH})_2(\text{bbta})$ (c), depicting the hexagonal framework pores and the local coordination environment of the cobalt centers in each material. Purple, red, blue, grey, and white spheres represent Co, O, N, C, and H atoms, respectively. Schemes illustrating NO binding to the cobalt(II) centers of $\text{Co}_2\text{Cl}_2(\text{bbta})$ (b) and $\text{Co}_2(\text{OH})_2(\text{bbta})$ (d). In the latter material, the more electron-rich cobalt centers reduce NO upon binding, generating cobalt(III)-NO⁻ adducts stabilized in part by a hydrogen bonding interactions with the framework bridging hydroxo groups, as indicated by spectroscopic and powder X-ray diffraction data. NO adsorption isotherms obtained for $\text{Co}_2\text{Cl}_2(\text{bbta})$ (green symbols) and $\text{Co}_2(\text{OH})_2(\text{bbta})$ (purple symbols) at 298 K, with pressure shown on a linear (e) and logarithmic (f) scale. Closed and open circles represent adsorption and desorption points, respectively.

Results and discussion

The frameworks $\text{Co}_2(\text{OH})_2(\text{bbta})$ and $\text{Co}_2\text{Cl}_2(\text{bbta})$ have previously been studied as adsorbents for O_2 ,^{40,41} and CO_2 ,⁴² as well as for the electrocatalytic oxidation of water to oxygen and photocatalytic reduction of CO_2 to CO .^{43,44} These two isostructural frameworks contain hexagonal pores lined with chains of coordinatively unsaturated cobalt centers bridged by triazolate and chloride or hydroxide groups (Fig. 1a). The solvent molecules that bind these cobalt centers can be removed by heating under vacuum, allowing the resulting, five-coordinate cobalt ions to interact with adsorbates and show promising behavior for many applications. In the case of O_2 binding, the strength of adsorption is dramatically enhanced when the framework bridging ligand is changed from chloride to hydroxide.^{40,41} This result is consistent with a more electron-rich ligand environment for the cobalt centers in $\text{Co}_2(\text{OH})_2(\text{bbta})$ relative to those in $\text{Co}_2\text{Cl}_2(\text{bbta})$, which enables them to more readily reduce O_2 , although hydrogen-bond donation from the hydroxide ligand to the resulting superoxide also contributes significantly.⁴⁵ Because NO, like O_2 , is capable of accepting an electron from a redox-active metal center,⁷ we sought to investigate how the electronic environment of the two frameworks may also give rise to distinct NO adsorption behavior.

Notably, NO adsorption isotherms collected for both materials at 298 K exhibit an initial steep rise at low pressures, indicative of strong framework–gas interactions (Fig. 1e). At 0.1 bar, both materials adsorb approximately 92% of their theoretical capacity, assuming one molecule of NO per cobalt center (5.30 and 6.02 mmol g^{-1} for $\text{Co}_2\text{Cl}_2(\text{bbta})$ and $\text{Co}_2(\text{OH})_2(\text{bbta})$, respectively; Fig. 1f). Both frameworks exhibit a much higher affinity for NO than O_2 ,⁴⁰ and the NO isotherm for $\text{Co}_2(\text{OH})_2(\text{bbta})$ is only slightly steeper than that of $\text{Co}_2\text{Cl}_2(\text{bbta})$ at low pressures. While the uptake of both frameworks begins to plateau above 120 mbar, the uptake of $\text{Co}_2(\text{OH})_2(\text{bbta})$ rises again to reach a capacity of 10.1 mmol g^{-1} (23.3 wt%) at 1 bar, one of the highest gravimetric capacities of any NO absorbent. This value can be contrasted to those previously observed, such as 6.5 mmol g^{-1} for $\text{Fe}_2(\text{dobdc})$,²⁸ ~ 7.2 mmol g^{-1} $\text{Ni}_2(\text{dobdc})$,²³ and 7.1 mmol g^{-1} for UiO-66-NH_2 .³³ Thus, its capacity corresponds to an unexpected 1.57 equiv. of NO per cobalt center, suggestive of additional reactivity beyond the binding of NO to the metal sites. In contrast, the NO capacity of $\text{Co}_2\text{Cl}_2(\text{bbta})$ is 6.6 mmol g^{-1} at 1 bar, which corresponds to 1.14 equiv. of NO gas per metal center. Interestingly, both frameworks also exhibit hysteresis upon desorption (Fig. 1f), although their recyclability of adsorption is drastically different: $\text{Co}_2\text{Cl}_2(\text{bbta})$ retains 97.4% of its adsorption capacity over three adsorption/desorption cycles with regeneration at 363 K, while the capacity of $\text{Co}_2(\text{OH})_2(\text{bbta})$ is greatly reduced after one cycle, even with attempted regeneration at temperatures up to 423 K (see the ESI and Fig. S1–S3† for details).

Nitric oxide uptake in both frameworks was further characterized by powder X-ray diffraction, infrared spectroscopy, and SQUID magnetometry. The powder X-ray diffraction pattern for

$\text{Co}_2\text{Cl}_2(\text{bbta})$ dosed with 2 mbar of NO gas at 298 K features changes in unit cell parameters and peak intensities relative to the desolvated framework, indicative of structural changes upon NO adsorption (Fig. S21 and S22†). Rietveld refinement yielded a structural model of NO-dosed $\text{Co}_2\text{Cl}_2(\text{bbta})$ ($\text{Co}_2\text{Cl}_2(\text{bbta})(\text{NO})_{1.59}$), wherein NO is bound in a bent, end-on fashion with a Co–N–O angle of $136.8(6)^\circ$, a Co–N_{NO} distance of 1.860(9) Å, and an N–O distance of 1.123(9) Å (Fig. 2a). The apparent compression of the N–O bond relative to that of free NO (1.154 Å) may be due to librational disorder. The unit cell volume of $\text{Co}_2\text{Cl}_2(\text{bbta})$ decreases only slightly by 98.3 Å³ (a 2.2% decrease) upon NO dosing, although the metal–ligand bonds increase slightly compared to those in the activated material. For example, two Co–N_{triazolate} bond distances increase from 2.064(10) and 2.106(6) Å in $\text{Co}_2\text{Cl}_2(\text{bbta})$ to 2.094(5) and 2.135(9) Å in the NO-dosed framework, while the Co–Cl bond length increases from 2.377(3) to 2.3920(17) Å. Thus, the decrease in unit cell volume can be attributed primarily to a decrease in the Cl–Co–Cl bond angle from $179.59(14)^\circ$ to $174.23(11)^\circ$.

Interestingly, the structural changes upon NO binding do not appear to be related to changes in the cobalt oxidation or spin state, as the cobalt–ligand bond lengths of the framework are consistent with those of high-spin cobalt(II).^{46–48} However, upon slowly cooling to 100 K (Fig. 2b), the occupancy of the

Fig. 2 Structural models determined from analysis of powder X-ray diffraction data for $\text{Co}_2\text{Cl}_2(\text{bbta})$ dosed with 2 mbar NO gas at 298 K (a) and 100 K (b), as well as $\text{Co}_2(\text{OH})_2(\text{bbta})$ dosed with 2 mbar (c) and 200 mbar NO gas (d) at 298 K. Purple, red, blue, green, and grey spheres represent Co, O, N, Cl, and C atoms respectively; H atoms are not shown as their positions could not be determined. In (a)–(c), the occupancy of the represented oxygen atom position is half of that of the NO-based nitrogen atom, and the position of a second oxygen atom with the same degree of occupancy (generated by symmetry) is not shown for clarity. For the structure of $\text{Co}_2(\text{OH})_2(\text{bbta})$ dosed with 200 mbar NO, the occupancy of the NO oxygen atom increases to 56.1(4)% of the NO nitrogen atom occupancy, nearly 15% more than would be expected for the structure with bound NO. As a result, it is indicative of the presence of both bound NO and NO_2^- species.

adsorbed NO species increases from 79.5(9)% to 91.8(12)%, concomitant with an elongation of the axial Co–N_{triazolate} bond (by 0.018 Å, to 2.153(6) Å) and a contraction of the equatorial Co–N_{triazolate} (by 0.031 Å, to 2.063(4) Å) and Co–Cl bonds (by 0.012 Å, to 2.380(3) Å). The changes in bond lengths suggest that the NO-bound cobalt centers may have partial low-spin character at low temperatures. Additionally, the elongation of the bond *trans* to the NO adduct is consistent with the strong *trans* influence of NO⁴⁹ observed in other compounds, such as nitrosyl-cobalamin and Fe₂(dobdc), where the metal–ligand bond *trans* to NO increases to more than 2.3 Å in length.^{18,28}

In contrast, dosing a sample of desolvated Co₂(OH)₂(bbta) with 2 mbar of NO gas at 298 K results in a much more drastic decrease in the framework unit cell volume, from 4427.7(3) to 4101.9(3) Å³, equivalent to a 7.3% contraction. Rietveld refinement against the diffraction data (Fig. 2c and S23†) led to a structural model with the formula Co₂(OH)₂(bbta)(NO)_{1.82} featuring a cobalt-bound NO and bent Co–N–O angle. As a result of a crystallographic two-fold axis along the Co–N_{NO} bond, the NO-based oxygen atom is disordered over two positions, each with 45.7(5)% occupancy. The structural parameters are indicative of the presence of a cobalt(III)–nitrosyl species (Fig. 2c).⁷ Specifically, the Co–N_{triazolate} distances decrease from 2.102(4) and 2.066(17) Å in the bare framework to 1.980(6) and 2.029(10) Å in the NO-dosed material, supporting either a spin-state change or charge transfer to generate cobalt(III). The slight elongation of the N–O bond distance to 1.134(16) Å, the Co–N_{NO} distance of 1.864(16) Å, and the Co–N–O angle of 128.5(11)° are also consistent with those observed for other six-coordinate {CoNO}⁸ systems.^{18,50,51} For example, characterization of nitrosyl-cobalamin derivatives *via* single-crystal X-ray diffraction revealed Co–N_{NO} bond lengths ranging from 1.907(2) to 1.940(8) Å and Co–N–O angles ranging from 118.9(8)° to 120.2(8)°.¹⁸ A lengthening of the metal–ligand bond *trans* to the NO is not observed for NO-dosed Co₂(OH)₂(bbta). The absence of such a perturbation in this case might be due to an inability of the highly contracted lattice of Co₂(OH)₂(bbta)(NO)_{1.82} to support substantial and asymmetric deviations in bond length. Finally, the NO-based oxygen atom is resolved at a distance of 3.079(19) Å from the oxygen atom of the nearest bridging hydroxo group, less than that of the combined crystallographic van der Waals radii of two oxygen atoms (3.1 Å).⁵² This proximity indicates that hydrogen bonding may contribute to the adsorption of NO at low pressures,^{40,41,44} as it is known to promote the reduction of NO to NO[−].⁵³

Powder X-ray diffraction data were also collected for Co₂(OH)₂(bbta) under 200 mbar of NO to investigate the structural origins of the additional gas uptake at higher pressures (Fig. 2d and S24†). Rietveld refinement against this data revealed a structural model wherein the occupancy of the NO oxygen increased from 50% to 60% of that of the NO nitrogen. This change, in tandem with the elongation of the N–O bond distance to 1.17(3) Å, suggests that the metal centers may be partially occupied by nitrite (NO₂[−]) as well as nitrosyl, owing to NO disproportionation upon gas uptake at higher pressures. In this structure, the NO oxygen is also much closer to the framework hydroxo group than in the 2 mbar structure, with an

O(H)⋯O_{NO} distance of 2.85(4) Å and a smaller Co–N–O angle of 116.0(19)° (Fig. 2d). These structural differences further indicate that at higher NO pressures, hydrogen bonding can significantly stabilize the adsorbed species in Co₂(OH)₂(bbta).

In situ infrared spectra were collected for NO-dosed samples of both frameworks to further elucidate changes upon NO adsorption. In the case of Co₂Cl₂(bbta), these data suggest that there is little electronic perturbation of the cobalt center upon NO binding, consistent with the diffraction data. For example, dosing with 1 mbar of NO gas results in the appearance of a peak at 1857 cm^{−1} that shifts to 1841 cm^{−1} under increasing pressures of NO up to 1 bar (Fig. S17†), energies that are only slightly different from the value for free NO (1876 cm^{−1}). Additionally, the bands associated with framework vibrations do not shift significantly upon NO binding (Fig. S16†), signifying that adsorption does not substantially perturb the framework structure. While it was not possible to heat the sample in the IR spectrometer to fully desorb NO after dosing with 1 bar, evacuation at 298 K over the course of eight hours resulted in a spectrum resembling that of the sample when dosed with 1 mbar of NO (Fig. S19†), indicative of some degree of reversible adsorption.

Like the gas adsorption and powder X-ray diffraction data, the infrared spectrum of Co₂(OH)₂(bbta) changes more dramatically upon dosing with NO, consistent with a change in the oxidation state of the cobalt and similar to what occurs upon dosing the material with O₂.⁴⁰ In order to deconvolute the bands arising due to NO adducts, infrared spectra were collected upon dosing with ¹⁴NO (up to 750 mbar) and ¹⁵NO (up to 176 mbar) at 298 K (Fig. 3a, b and S9†). Under 11 mbar of ¹⁴NO, two new broad features appear at 1814 and 1652 cm^{−1} and these shift to 1780 and 1624 cm^{−1}, respectively, when the framework is dosed with the same pressure of ¹⁵NO (Fig. S9†). These features were assigned as the ν(NO) band of multiple cobalt(III)–NO[−] adducts. At a pressure of 750 mbar ¹⁴NO, these bands shift to higher energies of 1844 and 1730 cm^{−1}, respectively, indicating less perturbation of the NO adduct. The appearance of multiple NO-derived species and their subsequent perturbation at higher loadings is consistent with earlier studies performed on this framework,⁴⁰ which indicated that the bridging nature of the ligands, as well as the close proximity of neighboring cobalt centers, lowered the reduction potential of cobalt(II) centers situated next to cobalt(III) sites that had been oxidized upon gas binding. As a result, a variety of bound superoxide species, depending upon O₂ loading, were observed. We propose an analogous phenomenon here, in which the cobalt(III)–NO[−] species generated upon binding of NO (ν_{NO} = 1652 cm^{−1}) results in a lower reduction potential of adjacent cobalt(II) centers. These cobalt(II) sites still bind NO (ν_{NO} = 1814 cm^{−1}), but are not able to reduce the gas to the same extent. As the electron density of the framework decreases at higher pressures due to increased oxidant loading and the formation of NO₂[−] species (see below), these bands further undergo a blue shift as both types of bound NO become less reduced. These less reduced species may not be as liable to disproportionate, contributing to their continued presence in the IR spectra. Additionally, we postulate that at 750 mbar, the band at

Fig. 3 (a) Raw DRIFTS spectra collected at 298 K for desolvated Co₂(OH)₂(bbta) under vacuum (black trace) and dosed with 1.4, 11, and 750 mbar ¹⁴NO gas (blue, teal, and green traces, respectively) between 2350 and 1500 cm⁻¹. This region features stretches that can be assigned to cobalt(III)–NO (1650–1840 cm⁻¹) and N₂O (2220 cm⁻¹). The difference spectra relative to the desolvated framework for each gas pressure are shown at the bottom of the figure. (b) Difference spectra for Co₂(OH)₂(bbta) dosed with 750 mbar ¹⁴NO gas (green trace) and 176 mbar ¹⁵NO gas (purple trace) relative to the desolvated framework under vacuum, between 800 and 1500 cm⁻¹. New bands at approximately 1460 and 1330 cm⁻¹ in the spectrum for the ¹⁴NO-dosed sample and approximately 1424 and 1300 cm⁻¹ in the spectrum for the ¹⁵NO-dosed sample are highlighted in green and purple, respectively, and can be distinguished from the non-highlighted framework shifts, which are consistent in the presence of both gases. We note that the data for ¹⁴NO and ¹⁵NO were collected using different pressures, which may account for the deviations in peak positions at lower wavenumbers. (c) Proposed scheme for NO uptake in Co₂(OH)₂(bbta), which proceeds first through initial binding of 1 equiv. of NO per cobalt center to form a cobalt(III)–nitrosyl species (NO pressures up to at least 11 mbar) followed by subsequent disproportionation in the presence of additional NO at higher pressures (176 or 200 mbar as indicated by the infrared and diffraction data, respectively) to form a cobalt(III)–bound NO₂⁻ and gaseous N₂O.

1844 cm⁻¹ may also represent some portion of NO bound to cobalt(III) sites that evolved N₂O by disproportionation and became coordinatively unsaturated, allowing them to interact with NO gas further by physisorption. The complexity of the infrared spectra, and the presence of multiple NO-derived species, is thus reasonable.

Additionally, at this higher pressure new bands appear at 1460 and 1326 cm⁻¹ (1424 and 1300 cm⁻¹, respectively, for the material dosed with 176 mbar ¹⁵NO), consistent with the asymmetric and symmetric NO stretching mode of an NO₂⁻ (nitro) moiety bound to cobalt(III) (Fig. 3b and S14†).^{54,55} Another new peak at 2220 cm⁻¹ corresponds to free N₂O (Fig. 3a and S11, S12†).⁵⁶ Due to poor scattering at lower wavenumbers, it was not possible to observe the NO₂ bend or Co–N stretch, which are predicted to appear at approximately 830 and 500 cm⁻¹.^{54,55,57} We further note that there is no distinct isotopically-sensitive NO band between 1100 and 1050 cm⁻¹, which would indicate the presence of a nitrito rather than a nitro moiety (*i.e.*, O rather than N-linkage).⁵⁷ This observation is in contrast to that recently noted in two copper(I)-based frameworks.^{34,35} Notable changes also occur to the ν(OH) stretch of the bridging hydroxo unit upon NO dosing: the band of the desolvated framework at 3647 cm⁻¹ diminishes substantially upon dosing with NO pressures of 750 mbar (¹⁴NO) or 176 mbar (¹⁵NO), while a new

set of bands appears at 3598 cm⁻¹ under 11 mbar of ^{14/15}NO (Fig. S8 and S10†). At the highest pressures examined for each gas, a second, even more redshifted band is present at 3517 cm⁻¹ (Fig. S8†), which was assigned to formation of a hydrogen bond between the bridging hydroxo group and the adsorbed NO (or NO₂⁻) species.

Dc magnetic susceptibility data were collected for Co₂Cl₂(bbta) and Co₂(OH)₂(bbta), both as desolvated and NO-dosed samples, to further probe changes occurring upon NO binding in each material. In the case of Co₂Cl₂(bbta), at 300 K the molar magnetic susceptibility temperature product (χ_MT) is 3.39 emu K mol⁻¹ per cobalt atom, in accord with an *S* = 3/2 (*g* = 2.69) complex (Fig. 4a). Upon dosing with one equiv. of NO, the value of χ_MT at 300 K decreases to 1.23 emu K mol⁻¹ per cobalt, in accord with an *S* = 1 (*g* = 2.22) complex. While the high degree of covalency in metal nitrosyl complexes complicates the assignment of formal oxidation states, and thus the interpretation of magnetic data,⁵⁰ the *S* = 1 state is likely best described as arising from strong antiferromagnetic coupling between a high-spin *S* = 3/2 cobalt(II) center and an *S* = 1/2 NO molecule. Upon cooling from approximately 160 to 120 K, a decrease in the magnetic susceptibility is observed. This may be attributed to cobalt(II)-based spin-crossover from *S* = 3/2 to *S* = 1/2, in agreement with the variable-temperature powder X-ray diffraction data (Fig. S28†). A

Fig. 4 (a) Plot of $\chi_M T$ versus T for desolvated $\text{Co}_2\text{Cl}_2(\text{bbta})$ (green symbols) and desolvated $\text{Co}_2\text{Cl}_2(\text{bbta})$ dosed with 1 equiv. of NO per cobalt (blue symbols). (b) Plot of $\chi_M T$ versus T for desolvated $\text{Co}_2(\text{OH})_2(\text{bbta})$ (yellow symbols) and desolvated $\text{Co}_2(\text{OH})_2(\text{bbta})$ dosed with 1 equiv. of NO per cobalt (purple symbols). All data were collected under a dc field of 0.1 T. Data for desolvated $\text{Co}_2\text{Cl}_2(\text{bbta})$ and $\text{Co}_2(\text{OH})_2(\text{bbta})$ are reproduced from ref. 40.

magnetic state arising from an uncoupled $S = 1/2$ ($g = 2.00$) cobalt(II) center and an $S = 1/2$ ($g = 2.00$) NO molecule gives an expected $\chi_M T$ value of $0.75 \text{ emu K mol}^{-1}$ per cobalt, which is near the $\chi_M T$ value of $0.86 \text{ emu K mol}^{-1}$ per cobalt at 120 K. Indeed, taken in concert with the gas adsorption, powder X-ray diffraction, and infrared spectroscopy data described above, NO appears to undergo only a minor perturbation upon binding to the metal sites of $\text{Co}_2\text{Cl}_2(\text{bbta})$, consistent with a lack of electron transfer from cobalt to NO.

Dc magnetic susceptibility data collected for NO-dosed $\text{Co}_2(\text{OH})_2(\text{bbta})$ are also distinct from data for the desolvated framework (Fig. 4b). At 300 K, the desolvated framework has a $\chi_M T$ value of $2.42 \text{ emu K mol}^{-1}$ per cobalt atom, corresponding to an $S = 3/2$ ($g = 2.27$) complex. The increased ligand field strength of the hydroxide ligand set is likely responsible for a decreased orbital contribution to the magnetic moment and a lower g -value relative to the desolvated $\text{Co}_2\text{Cl}_2(\text{bbta})$ framework.⁴⁸ Dosing $\text{Co}_2(\text{OH})_2(\text{bbta})$ with one equiv. of NO resulted in a significant decrease in $\chi_M T$ at 300 K, from 2.42 to $0.575 \text{ emu K mol}^{-1}$ per cobalt. At 300 K, the moment of the NO-dosed framework is lower

than the value predicted for both a coupled or uncoupled high-spin $S = 3/2$ cobalt(II) and an $S = 1/2$ NO unit (1 and $2.25 \text{ emu K mol}^{-1}$, respectively) as well as that predicted for an $S = 0$ cobalt(III) center and an $S = 1$ NO^- (1 emu K mol^{-1}). Measurements at multiple applied fields reveal no evidence for temperature-independent paramagnetism or ferromagnetic impurities (Fig. S32†). Taken together with the clear shift of the $\nu(\text{NO})$ and framework-based bands in the infrared spectroscopy data and the changes in structural parameters of both the framework and NO adduct in the powder X-ray diffraction data, the magnetic data generally support the reduction of NO by $\text{Co}_2(\text{OH})_2(\text{bbta})$ upon adsorption. Indeed, the lower than expected moment may be due to further disproportionation of cobalt(III)-bound NO^- to form cobalt(III)-bound NO_2^- and N_2O gas (see below).

Overall, the foregoing data reveal that adsorption of NO in $\text{Co}_2(\text{OH})_2(\text{bbta})$ is accompanied by more drastic structural and electronic changes than in the case of $\text{Co}_2\text{Cl}_2(\text{bbta})$. These changes are indicative of electron transfer from cobalt(II) to NO in $\text{Co}_2(\text{OH})_2(\text{bbta})$ and are consistent with the more basic ligand set in this framework relative to $\text{Co}_2\text{Cl}_2(\text{bbta})$, which can stabilize cobalt(III) over cobalt(II).⁵⁸ Significantly, as evidenced by infrared spectroscopy data at pressures above 176 mbar and diffraction data obtained at 200 mbar NO, charge transfer to adsorbed NO appears to further facilitate some degree of disproportionation to form cobalt(III)-bound NO_2^- and N_2O gas. In a gas adsorption isotherm measurement, the formed N_2O is likely to desorb into the headspace of the sample holder such that, for a given pressure, the equivalency of NO as measured by the gas adsorption analyzer will appear less than the actual NO that is required to generate the corresponding proportion of cobalt(III)- NO_2^- adducts. Taking this into account, and assuming that at 1 bar of NO the adsorbed species is exclusively NO_2^- , a capacity of 10.1 mmol g^{-1} (1.57 equiv. NO per Co, Fig. 1e and f) would suggest as many as 78.5% of the cobalt sites have reacted with coordinated NO to form bound NO_2^- and gaseous N_2O , a similar level of oxidative reactivity to that observed previously with O_2 .³³

The occurrence of a disproportionation reaction at higher NO pressures also explains why $\text{Co}_2(\text{OH})_2(\text{bbta})$ cannot be fully regenerated after NO uptake at pressures beyond a few hundred mbar. Indeed, attempted regeneration of a sample of $\text{Co}_2(\text{OH})_2(\text{bbta})$ following collection of adsorption data yielded a crystalline solid with a powder X-ray diffraction pattern distinct from that of desolvated $\text{Co}_2(\text{OH})_2(\text{bbta})$ (Fig. S27†). In particular, the resulting unit cell volume is contracted relative to that of the parent desolvated material, which indicates that the material remains partially oxidized, as would be expected upon formation of cobalt(III)-bound NO_2^- and N_2O gas. While a satisfactory Rietveld refinement could not be obtained, Fourier difference maps suggest residual electron density over the metal centers and near the hydroxo group. These data are suggestive of bound NO_2^- species that would likely display strong interactions with the bridging hydroxo groups. Additionally, these differences are consistent with those observed in the infrared spectra of the framework upon NO dosing at high pressures, such as the generation of a cobalt(III)-nitro species

and dramatic shifts in the hydroxo group bands. Thus, hydrogen bonding likely plays an important role in facilitating NO disproportionation, much like with other examples of nearby Lewis acids and hydrogen bond donors aiding in transition metal complex reactivity with NO.^{59–61} In particular, the hydrogen bond may help to further polarize the NO adduct, promoting its reaction with additional equivalents of nitric oxide to form a putative Co(N₂O₂) species. We postulate that the cobalt centers are then close enough across the pore to enable Co(N₂O₂) and Co(NO) species to interact and disproportionate, generating N₂O gas and cobalt(III)-bound NO₂[−] (Fig. S30†).

Conclusions

The foregoing results demonstrate that both Co₂Cl₂(bbta) and Co₂(OH)₂(bbta) are capable of strongly binding NO, and notably these materials constitute only the second and third examples of cobalt-based metal–organic frameworks for NO capture. In particular, Co₂Cl₂(bbta) displays reversible nitric oxide binding and a high adsorption capacity at room temperature, making it a promising material for nitric oxide delivery. Additionally, data collected on NO-dosed Co₂Cl₂(bbta) at low temperatures suggest spin-crossover behavior to afford low-spin NO-bound cobalt(II). We note that while data on the toxicity of Co₂Cl₂(bbta) and Co₂(OH)₂(bbta) are unavailable, one study that examined the effects of a wide range of metal–organic frameworks in human cell lines and zebrafish embryo found that the cobalt-based framework Co₂(dobdc) showed minimal toxicity *in vitro* and *in vivo*, and that health effects of frameworks *in vitro* and *in vivo* appeared to be related to metal ion identity.⁶² These results suggest that Co₂Cl₂(bbta) might be similarly benign.

The introduction of a more basic bridging hydroxo ligand in the material Co₂(OH)₂(bbta) promotes charge transfer from cobalt(II) to NO, and at higher pressures (>0.1 bar) of NO, disproportionation is clearly favored, with formation of N₂O and generation of cobalt(III)-nitro species, as observed by *in situ* infrared spectroscopy. Additionally, powder X-ray diffraction and spectroscopic data indicate that the hydroxo moiety of the framework engages in hydrogen bonding to the adsorbed NO, which seemingly promotes the observed disproportionation reactivity at higher pressures. As in the case of Fe-MOF-5 and Cu-MFU-4l, it is anticipated that further study of NO adsorption in this material at lower temperatures, as well as computational validation, could elucidate the identities of intermediates of NO disproportionation (*i.e.*, Co(N₂O₂)), to corroborate the mechanism of NO disproportionation in this material. This knowledge would be useful for tuning and controlling denitrification reactions within metal–organic frameworks,^{37,63} and future studies could contribute critical insights in this regard.^{21,34,35,64–67}

Data availability

Full experimental details, supplementary figures and tables, gas sorption data, refinement information from powder X-ray diffraction analyses, infrared spectra, and magnetometry data are provided in the ESI. CCDC: 2098007–2098011.†

Author contributions

J. O. and J. R. L. formulated the project. J. O. synthesized the materials and measured and analyzed the gas adsorption data. J. O. and H. Z. H. J. collected the powder X-ray diffraction data, and J. O. analyzed the data. H. Z. H. J. collected and analyzed the infrared spectra, with contributions from J. O. A. B. T. collected and analyzed the magnetic susceptibility data. J. O. and J. R. L. wrote the paper, and all authors contributed to revising the paper.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 E. Culotta and D. E. Koshland, *Science*, 1992, **258**, 1861–1865.
- 2 L. J. Ignarro, G. M. Buga, K. S. Wood, R. E. Byrns and G. Chaudhuri, *Proc. Natl. Acad. Sci. U. S. A.*, 1987, **84**, 9265–9269.
- 3 M. R. Miller and I. L. Megson, *Br. J. Pharmacol.*, 2007, **151**, 305–321.
- 4 J. Martel, Y.-F. Ko, J. D. Young and D. M. Ojcius, *Microbes Infect.*, 2020, **22**, 168–171.
- 5 L. Chen, P. Liu, H. Gao, B. Sun, D. Chao, F. Wang, Y. Zhu, G. Hedenstierna and C. G. Wang, *Clin. Infect. Dis.*, 2004, **39**, 1531–1535.
- 6 S. Åkerström, V. Gunalan, C. T. Keng, Y. J. Tan and A. Mirazimi, *Virology*, 2009, **395**, 1–9.
- 7 J. A. McCleverty, *Chem. Rev.*, 2004, **104**, 403–418.
- 8 J. A. McCleverty, *Chem. Rev.*, 1979, **79**, 53–76.
- 9 A. R. Butler and I. L. Megson, *Chem. Rev.*, 2002, **102**, 1155–1165.
- 10 R. Radi, *Chem. Res. Toxicol.*, 1996, **9**, 828–835.
- 11 S. H. Francis, J. L. Busch and J. D. Corbin, *Pharmacol. Rev.*, 2010, **62**, 525–563.
- 12 G. C. Brown, *Biochim. Biophys. Acta, Bioenerg.*, 2001, **1504**, 46–57.

- 13 R. Gardiner, T. G. Traylor, V. S. Sharma and H. Mizukami, *Biochemistry*, 1987, **26**, 3837–3843.
- 14 J. F. Deatherage and K. Moffat, *J. Mol. Biol.*, 1979, **134**, 401–417.
- 15 M. Brouwer, W. Chamulitrat, G. Ferruzzi, D. L. Sauls and J. B. Weinberg, *Blood*, 1996, **88**, 1857–1864.
- 16 L. Hannibal, C. A. Smith, D. W. Jacobsen and N. E. Brasch, *Angew. Chem., Int. Ed.*, 2007, **46**, 5140–5143.
- 17 M. Wolak, A. Zahl, T. Schnepf, G. Stochel and R. Van Eldik, *J. Am. Chem. Soc.*, 2001, **123**, 9780–9791.
- 18 H. A. Hassanin, M. F. El-Shahat, S. DeBeer, C. A. Smith and N. E. Brasch, *Dalton Trans.*, 2010, **39**, 10626.
- 19 A. C. McKinlay, R. E. Morris, P. Horcajada, G. Férey, R. Gref, P. Couvreur and C. Serre, *Angew. Chem., Int. Ed.*, 2010, **49**, 6260–6266.
- 20 S. Zhang, M. M. Melzer, S. N. Sen, N. Çelebi-Ölçüm and T. H. Warren, *Nat. Chem.*, 2016, **8**, 663–669.
- 21 P. Tavares, A. S. Pereira, J. J. G. Moura and I. Moura, *J. Inorg. Biochem.*, 2006, **100**, 2087–2100.
- 22 R. V. Pinto and M. L. Pinto, in *Therapeutic Application of Nitric Oxide in Cancer and Inflammatory Disorders*, Elsevier, 2019, pp. 277–305.
- 23 N. J. Hinks, A. C. McKinlay, B. Xiao, P. S. Wheatley and R. E. Morris, *Microporous Mesoporous Mater.*, 2010, **129**, 330–334.
- 24 T. Yang, A. N. Zelikin and R. Chandrawati, *Adv. Sci.*, 2018, **5**, 1701043.
- 25 H. C. Zhou, J. R. Long and O. M. Yaghi, *Chem. Rev.*, 2012, **112**, 673–674.
- 26 H. Furukawa, K. E. Cordova, M. O’Keeffe and O. M. Yaghi, *Science*, 2013, **341**, 1230444.
- 27 B. Xiao, P. S. Wheatley, X. Zhao, A. J. Fletcher, S. Fox, A. G. Rossi, I. L. Megson, S. Bordiga, L. Regli, K. M. Thomas and R. E. Morris, *J. Am. Chem. Soc.*, 2007, **129**, 1203–1209.
- 28 E. D. Bloch, W. L. Queen, S. Chavan, P. S. Wheatley, J. M. Zadrozny, R. Morris, C. M. Brown, C. Lamberti, S. Bordiga and J. R. Long, *J. Am. Chem. Soc.*, 2015, **137**, 3466–3469.
- 29 F. Bonino, S. Chavan, J. G. Vitillo, E. Groppo, G. Agostini, C. Lamberti, P. D. C. Dietzel, C. Prestipino and S. Bordiga, *Chem. Mater.*, 2008, **20**, 4957–4968.
- 30 A. C. McKinlay, B. Xiao, D. S. Wragg, P. S. Wheatley, I. L. Megson, R. E. Morris, A. C. McKinlay, B. Xiao, D. S. Wragg, P. S. Wheatley, I. L. Megson and R. E. Morris, *J. Am. Chem. Soc.*, 2008, **130**, 10440–10444.
- 31 D. Cattaneo, S. J. Warrender, M. J. Duncan, C. J. Kelsall, M. K. Doherty, P. D. Whitfield, I. L. Megson and R. E. Morris, *RSC Adv.*, 2016, **6**, 14059–14067.
- 32 D. Cattaneo, S. J. Warrender, M. J. Duncan, R. Castledine, N. Parkinson, I. Haley and R. E. Morris, *Dalton Trans.*, 2016, **45**, 618–629.
- 33 R. R. Haikal, C. Hua, J. J. Perry, D. O’Nolan, I. Syed, A. Kumar, A. H. Chester, M. J. Zaworotko, M. H. Yacoub and M. H. Alkordi, *ACS Appl. Mater. Interfaces*, 2017, **9**, 43520–43528.
- 34 A. M. Wright, C. Sun and M. Dinca, *J. Am. Chem. Soc.*, 2021, **143**, 681–686.
- 35 C. Sun, L. Yang, M. A. Ortuño, A. M. Wright, T. Chen, A. R. Head, N. López and M. Dinca, *Angew. Chem., Int. Ed.*, 2021, 2–8.
- 36 A. C. McKinlay, J. F. Eubank, S. Wuttke, B. Xiao, P. S. Wheatley, P. Bazin, J.-C. Lavalley, M. Daturi, A. Vimont, G. De Weireld, P. Horcajada, C. Serre and R. E. Morris, *Chem. Mater.*, 2013, **25**, 1592–1599.
- 37 Z. Cai, W. Tao, C. E. Moore, S. Zhang and C. R. Wade, *Angew. Chem., Int. Ed.*, 2021, 2–7.
- 38 C. K. Brozek, J. T. Miller, S. A. Stoian and M. Dinca, *J. Am. Chem. Soc.*, 2015, **137**, 7495–7501.
- 39 C. K. Brozek and M. Dinca, *J. Am. Chem. Soc.*, 2013, **135**, 12886–12891.
- 40 J. Oktawiec, H. Z. H. Jiang, J. G. Vitillo, D. A. Reed, L. E. Darago, B. A. Trump, V. Bernales, H. Li, K. A. Colwell, H. Furukawa, C. M. Brown, L. Gagliardi and J. R. Long, *Nat. Commun.*, 2020, **11**, 3087.
- 41 A. S. Rosen, M. R. Mian, T. Islamoglu, H. Chen, O. K. Farha, J. M. Notestein and R. Q. Snurr, *J. Am. Chem. Soc.*, 2020, **142**, 4317–4328.
- 42 P. Q. Liao, H. Chen, D. D. Zhou, S. Y. Liu, C. T. He, Z. Rui, H. Ji, J. P. Zhang and X. M. Chen, *Energy Environ. Sci.*, 2015, **8**, 1011–1016.
- 43 X. F. Lu, P. Q. Liao, J. W. Wang, J. X. Wu, X. W. Chen, C. T. He, J. P. Zhang, G. R. Li and X. M. Chen, *J. Am. Chem. Soc.*, 2016, **138**, 8336–8339.
- 44 Y. Wang, N. Y. Huang, J. Q. Shen, P. Q. Liao, X. M. Chen and J. P. Zhang, *J. Am. Chem. Soc.*, 2018, **140**, 38–41.
- 45 R. D. Jones, D. A. Summerville and F. Basolo, *Chem. Rev.*, 1979, **79**, 139–179.
- 46 H. A. Goodwin, in *Topics in Current Chemistry*, 2004, pp. 23–47.
- 47 P. Thuery, J. Zarembowitch, A. Michalowicz and O. Kahn, *Inorg. Chem.*, 1987, **26**, 851–855.
- 48 Z. Dori and H. B. Gray, *Inorg. Chem.*, 1968, **7**, 889–892.
- 49 T. G. Traylor and V. S. Sharma, *Biochemistry*, 1992, **31**, 2847–2849.
- 50 J. H. Enemark and R. D. Feltham, *Coord. Chem. Rev.*, 1974, **13**, 339–406.
- 51 Here is referenced notation developed by Enemark and Feltham (50) for NO-bound transition metal complexes, given that NO is a non-innocent ligand capable of multiple possible oxidation states. Specifically, $\{M(NO)_x\}^n$, where M is the element, n is the sum of the initial metal d and the NO π^* orbitals, and x is the number of NO ligands.
- 52 S. S. Batsanov, *Inorg. Mater.*, 2001, **37**, 871–885.
- 53 M. D. Bartberger, W. Liu, E. Ford, K. M. Miranda, C. Switzer, J. M. Fukuto, P. J. Farmer, D. A. Wink and K. N. Houk, *Proc. Natl. Acad. Sci. U. S. A.*, 2002, **99**, 10958–10963.
- 54 K. Nakamoto, J. Fujita and H. Murata, *J. Am. Chem. Soc.*, 1958, **80**, 4817–4823.
- 55 R. B. Penland, T. J. Lane and J. V. Quagliano, *J. Am. Chem. Soc.*, 1956, **78**, 887–889.
- 56 R. L. Hudson, M. J. Loeffler and P. A. Gerakines, *J. Chem. Phys.*, 2017, **146**, 1–9.

- 57 K. Nakamoto, *Infrared and Raman Spectra of Inorganic and Coordination Compounds*, 2009, 1–273.
- 58 E. C. Niederhoffer, J. H. Timmons and A. E. Martell, *Chem. Rev.*, 1984, **84**, 137–203.
- 59 E. G. Abucayon, R. Khade, D. R. Powell, Y. Zhang and G. B. Richter-Addo, *Angew. Chem., Int. Ed.*, 2019, 2–8.
- 60 C. H. Chuang, W. F. Liaw and C. H. Hung, *Angew. Chem., Int. Ed.*, 2016, **55**, 5190–5194.
- 61 G. B. Wijeratne, M. Bhadra, M. A. Siegler and K. D. Karlin, *J. Am. Chem. Soc.*, 2019, **141**, 17962–17967.
- 62 A. Ruyra, A. Yazdi, J. Espín, A. Carné-Sánchez, N. Roher, J. Lorenzo, I. Imaz and D. Maspoch, *Chem.–Eur. J.*, 2015, **21**, 2508–2518.
- 63 X. Han, S. Yang and M. Schröder, *Nat. Rev. Chem.*, 2019, **3**, 108–118.
- 64 J. Jover, C. K. Brozek, M. Dincă and N. López, *Chem. Mater.*, 2019, **31**, 8875–8885.
- 65 A. Shiotari, S. Hatta, H. Okuyama and T. Aruga, *Chem. Sci.*, 2014, **5**, 922–926.
- 66 N. Yeung, Y. W. Lin, Y. G. Gao, X. Zhao, B. S. Russell, L. Lei, K. D. Miner, H. Robinson and Y. Lu, *Nature*, 2009, **462**, 1079–1082.
- 67 S. Matsumoto, K. Yokota, H. Doi, M. Kimura, K. Sekizawa and S. Kasahara, *Catal. Today*, 1994, **22**, 127–146.