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# Thermal expansion properties of organic crystals: a CSD study†

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The thermal expansion properties of crystalline organic compounds are investigated by data mining of the Cambridge Structural Database (CSD). The mean volumetric thermal expansion coefficient is  $168.8 \times 10^{-6} \text{ K}^{-1}$  and the mean uniaxial thermal expansion coefficient is  $71.4 \times 10^{-6} \text{ K}^{-1}$ , based on 745 and 1129 different observations, respectively. Normal and anomalous coefficients can be identified using these values and the associated standard deviations. The anisotropy of the thermal expansion is also evaluated and found to have a very broad distribution. 4719 different structures, comprising 4093 different molecular compounds and 626 additional polymorphs have been analyzed on their thermal expansion properties. Approximately 34% of these structures may have at least one orthogonal axis with negative thermal expansion, much more than generally believed. Moreover 127 structures have been identified which could have negative volumetric thermal expansion. Experimental validation using a robust protocol with data collected at more than 2 different temperatures is required to validate these cases.

## Introduction

Thermal expansion is the shape response of a material exposed to a temperature change. For solid materials the shape response is usually positive with positive temperature change, *i.e.* the material expands on heating. The isobaric volumetric thermal expansion coefficient expresses the change in volume with temperature per unit volume:

$$\alpha_V = \frac{1}{V} \left( \frac{\partial V}{\partial T} \right)_P$$

Likewise, the isobaric uniaxial thermal expansion coefficient expresses the change along a certain direction with temperature per unit length:

$$\alpha_L = \frac{1}{L} \left( \frac{\partial L}{\partial T} \right)_P$$

The units of the coefficients are in  $\text{K}^{-1}$  but because of its typical magnitudes they are rather expressed in  $10^{-6} \text{ K}^{-1}$ , or, more conveniently, in  $\text{MK}^{-1}$ . Uniaxial does not necessarily mean along a crystallographic axis, since the expansion can be calculated along any direction in space. Thermal expansion is supposed to be mostly linear, but need not be so.

Thermal expansion can be measured using mechanical dilatometry, optical methods such as optical interferometry, and by diffraction techniques.<sup>1</sup> However, thermal expansion coefficients measured by dilatometry sometimes give different values than those expected from diffraction measurements. This is especially true for polycrystalline materials with micro-cracks and microvoids, where the microscopic negative expansion dominates the atomic expansion.<sup>2</sup> IR spectroscopy has also been used recently for the measurement of thermal expansion properties of organic semiconducting single crystals.<sup>3</sup>

Modern diffraction techniques coupled with widely available cryogenic and high-temperature devices have made possible the measurement of thermal expansion with relative ease, provided that the material is available in crystalline form, *i.e.* either as single crystals or as crystalline powders. Despite these possibilities, systematic studies related to the thermal expansion properties are relatively rare, most possibly due to the fact that the atomic structure is usually the focus of the study. Thermal expansion properties play also a role in crystal structure prediction. It was shown that the errors introduced by neglecting the thermal expansion properties may affect the free-energy among polymorphs which are energetically close.<sup>4</sup> Another field where knowledge of volume and density plays a key role is volume-based thermodynamics.<sup>5,6</sup>

In the last few decades there is a growing interest in materials with anomalous shape properties related to temperature, pressure or strain stimuli. The focus is here on materials with negative or close to zero coefficients, the latter being in particular interesting for numerous applications where shape changes with temperature are undesirable.<sup>7–11</sup> The focus is usually on inorganic or hybrid materials that often have

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nanoporous structures, such as zeolites and metal–organic frameworks (MOFs), which have the advantage of being very flexible with temperature, pressure or uptake of guest species. Uniaxial negative thermal expansion or even volumetric negative thermal expansion have been reported in a number of these structures.<sup>12–16</sup> Technological applications require a fine degree of control of the thermal expansion, and this can be achieved through various means, chemical or mechanical, such as the use of pressure.<sup>17,18</sup> Devices composed of more than one material and operated in a range of temperatures could show degraded performances when the thermal expansion coefficients do not match. The starting point for the research of controllable thermal expansion is the study of materials having negative thermal expansion.<sup>19</sup> Dove & Fang have shown theoretically that negative thermal expansion becomes larger (more negative) with increasing pressure and positive thermal expansion less positive.<sup>20</sup> This theoretical behavior is sometimes observed,<sup>21</sup> sometimes not.<sup>22–24</sup> Chemical modification by intercalation can be used to control the thermal expansion, *e.g.* by reversible wetting in nanopores of MOFs,<sup>25</sup> adsorption of CO<sub>2</sub> in a MOF structure to reach a zero expansion material,<sup>26</sup> or by polymerizing ethylene in the pores of silicalite, which changes the sign of the volumetric thermal expansion from negative to positive.<sup>27</sup> More general design strategies to tune the thermal expansion properties in MOFs by chemical modification have been given recently by Burtch *et al.*<sup>28</sup> and theoretical considerations for these materials have been presented by Sanson.<sup>18</sup> General observations to obtain zero expansion functional materials by chemical modification have been given by Chen *et al.*<sup>29</sup>

Most of the examples of anomalous and/or tunable thermal expansion properties concern inorganic or hybrid organic/inorganic structures, such as metal–organic framework structures.<sup>2,30</sup> Less focus has been given to anomalous thermal expansion properties of organic structures. Some examples of tailored thermal expansion in organic structures include the study by Engel *et al.*<sup>31</sup> who show how the expansion can be tuned by chemical replacement of certain guest species in organic inclusion compounds and the negative thermal expansion in the COV-102 structure.<sup>32</sup> An exceptional negative volumetric thermal expansion in 1,4-diazabicyclo[2.2.2]octane hydrochloride was reported by Szafranski (2013);<sup>33</sup> the highest reported negative uniaxial thermal expansion was found in chloranilic acid pyrazine.<sup>34</sup>

In this paper we focus on thermal expansion properties of crystalline organic materials determined by diffraction techniques, using data mining of the Cambridge Structural Database (CSD).<sup>35</sup> It is shown that unusual thermal expansion properties exist for a number of crystalline compounds which have not been recognized as such before. The CSD contains at the time of writing somewhat more than one million of entries containing the results of diffraction experiments under a variety of conditions, and thus also possibly unexplored thermal expansion properties of crystalline materials. Crystal structures have been occasionally measured at different temperatures, either for one study by one research group, or by different research groups for unrelated studies. A CSD survey on density

properties of organic compounds as a function of temperature has been published in 2007 on the 2004 database version with 325 709 entries by a manual search and treatment in an external spreadsheet program.<sup>36</sup> In total 373 unique structures were found with temperature dependent data. The CSD Python API<sup>37</sup> gives the possibility to extract the data in a more systematic way and to treat the results numerically and statistically. For the present study the 5.41 version of the CSD was used.

## Extracting thermal expansion coefficients from the CSD

### Temperature data in CSD entries

**Basic CSD keys.** Each crystal structure reported in the CSD ('CSD entry') is labeled by an identifier (refcode) having at least six uppercase characters from the 26-letter Latin alphabet. A new measurement of the same chemical compound has the same refcode with a postfix consisting of an additional number starting at 01. The new measurement may be having performed under the same experimental conditions as the primary refcode measurement, or under different conditions (different temperature, pressure, exposure to UV, *etc.*). It is also possible that the new measurement was done on a polymorphic variant of the primary structure. If this is recognized by the CCDC, the entry is attributed a 'polymorph' key. The polymorph key is, however, not always uniquely defined, even it concerns in reality the same polymorph (*e.g.* the  $\beta$ -glycine polymorph). Sometimes polymorphism is not recognized, which could lead to erroneous attributions in thermal expansion coefficients. One case is refcode MEBVOH with two reported structures at two different temperatures. The publication reports two 'polyforms' MEBVOH and MEBVOH1 with the same space group and very close cell parameters.<sup>38</sup> The two structures are most probably in reality the two enantiomeric forms of the same molecule, but can be easily overlaid without inversion using the MERCURY program.<sup>39</sup> Since the atomic displacement parameters of the two structures were in the normal range for the reported temperatures, they were retained for the present analysis.

**Temperature information.** Each entry has also a 'temperature' key, which could be empty if the corresponding cif tag is empty or not defined and if there is no indication in the original publication at which temperature the experiment was performed. Sometimes the CCDC attributes 'room temperature' to the temperature key if there are anyhow indications that the experiments were performed at room temperature. In a number of cases the reported temperature is clearly wrong, such as in the case of ZEYVAA10,<sup>40</sup> which is a redetermination of the structure of the structure of ZEYVAA using the same experimental data.<sup>41</sup> In the case of ZEYVAA the reported temperature is 173 K, whereas for ZEYVAA10 it is room temperature, which obviously proves that the same authors have forgotten to correct the cif tag of the temperature for the redetermination. Temperatures are mostly reported in K, but entries reported in degrees Celsius are not rare, especially for older ones. In a number of cases the temperature key is absent, but temperature information could anyhow be recovered *via* the cif file in



the ‘\_diffrn\_ambient\_temperature’ tag. The temperature string of the CSD entry could not be decoded in all cases with confidence, and these were not taken into consideration for further processing.

**Curation of temperature information.** The structural information itself can be used as internal check for the data in order to know whether the reported temperature is trustworthy or not. Atomic displacement parameters,  $\langle U_{eq} \rangle_a$ , are directly correlated to the temperature of the measurement and are expected to be in certain reasonable ranges. Fig. 1 shows a scatterplot of the structure  $\langle U_{eq} \rangle_s$  value obtained by averaging the atomic  $\langle U_{eq} \rangle_a$  values of all non-hydrogen atoms, with full site occupancy, whereby excluding structures obtained by neutron diffraction and powder diffraction. For some entries, positional atomic information was present, but not the site occupancies. These structures have been retained for the analysis, assuming that the site occupancy for all atoms is 1.0. Only organic structures with reported  $R_F$  values below 5% were taken into account. It should be noted that structures have been reported with  $\langle U_{eq} \rangle_s$  as high as  $50 \text{ \AA}^2$  or as low as  $-0.4 \text{ \AA}^2$ . Therefore, structures with obviously erroneous ADP data were excluded from the plot when  $\langle U_{eq} \rangle_s < 0$  or  $\langle U_{eq} \rangle_s > 0.4 \text{ \AA}^2$ . Although there is considerable scatter in the plot, a clear positive correlation of  $\langle U_{eq} \rangle_s$  with temperature can be observed. A linear fit has a goodness of fit parameter  $R^2$  of 0.759. By binning the data in 10 K temperature intervals, an outlier criterion based on the interquartile range (IQR) method can be defined for the data in each bin.<sup>42</sup> In this way a maximum and minimum  $\langle U_{eq} \rangle_s$  can be determined for each temperature – with are set at  $1.5 \times \text{IQR}$  – in between which the  $\langle U_{eq} \rangle_s$  value should normally fall. For the analysis of

thermal expansion properties, structures were rejected with  $\langle U_{eq} \rangle_s$  values outside this temperature-dependent interval. There may be of course structures to which the average behavior does not apply and that are thus rejected unjustified, but it is felt that this method captures more cases with erroneous reported temperatures. Around 20 K the scatterplot shows a significant proportion of data points around  $0.015 \text{ \AA}^2$ , but also many values at much larger values, which can safely be attributed to measurements performed around  $20^\circ\text{C}$  instead of 20 K.

For temperatures beyond room temperature this ADP analysis method becomes rapidly less reliable, because of the scarcity of experimental observations.

### Space group and unit cell information in CSD entries

**Choice of structures.** For the analysis of thermal expansion properties, all organic structures with  $R_F$  below 10% were taken into account, including those from neutron and powder diffraction. The rationale for the higher  $R_F$  cutoff compared to that of the ADP analysis, is that in the latter case the results from the structural refinement are used which are directly connected to  $R_F$ . The precision of the cell parameters is not related to  $R_F$  but to a cell parameter least-squares  $R$  factor which is usually not reported. In this way a list of compounds was created with structures measured at least two reported temperatures. Only series of structures were taken into account with at least 50 K difference between the lowest and highest reported temperature.

**Space group and cell parameter consistencies.** Polymorphic compounds and those with a reported space group change were included in the data set whenever it was possible to isolate the

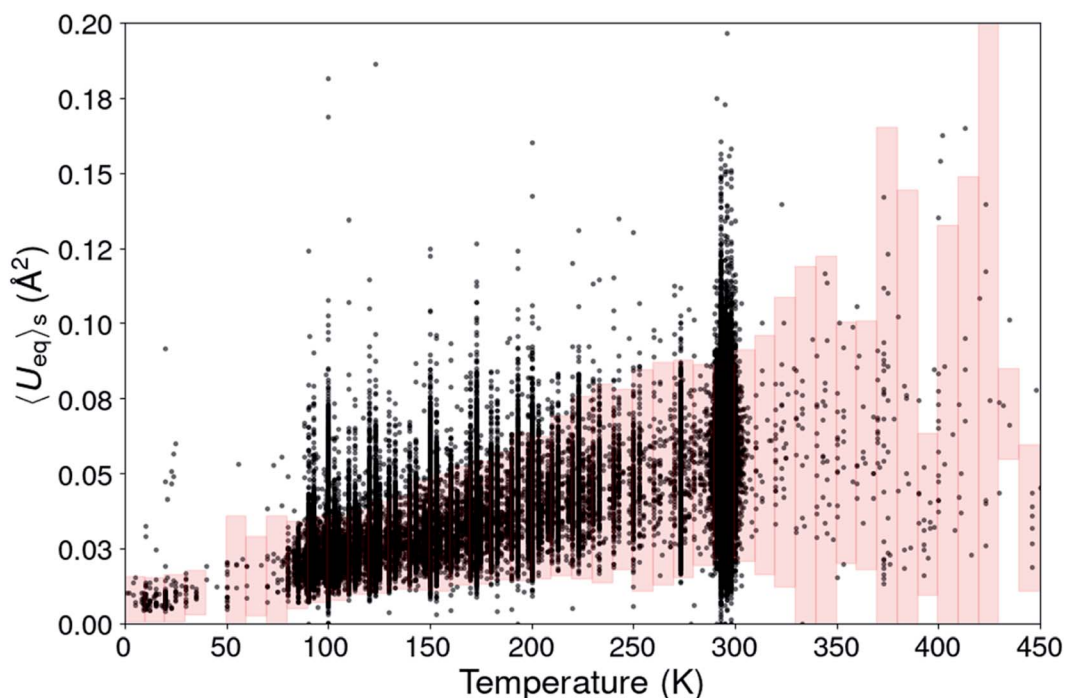


Fig. 1 Scatterplot of the structural  $\langle U_{eq} \rangle_s$  value from 180 672 organic CSD entries. The pink boxes define the  $\langle U_{eq} \rangle_s$  range for each temperature in which the  $\langle U_{eq} \rangle_s$  values are supposed to be.



structures reported in one space group and then treated as separate structures. For alternative settings of monoclinic space groups the cell parameters were all transformed to the conventional setting<sup>43</sup> of the international tables for crystallography using the REDUCE routine in the CRYSCALC library.<sup>44,45</sup> Cell parameters of orthorhombic space groups were transformed in such a way that  $a < b < c$ . Rhombohedral space groups were all transformed to the hexagonal setting. Despite these precautions still peculiar cases remained with large cell parameter jumps for subsequent temperatures; therefore, additional tests were carried out to verify if polymorphs were not overseen or simply that the reported symmetry was wrong. Since it was difficult to find optimal tolerance settings using a comparison of reduced cells using the methodology described by Andrews *et al.*,<sup>46</sup> a more basic cell comparison method was used with a tolerance of 0.5 Å for the axes and 2° for the angles. Any individual structure was discarded from the temperature range whenever its cell parameters did not match within tolerances with its predecessor in the temperature range, which was repeated iteratively until the series was homogeneous or when only one structure was left, in which case this structure was discarded as well. Cell parameters of structures with close temperatures were also merged within bins of 10 K, after which the temperature of the midpoint of the bin was attributed to the structure.

**CSD temperature check.** Finally structures with  $\langle U_{eq} \rangle_s$  outside the intervals defined for their temperature bin were rejected for the analysis. Note that this procedure cannot guarantee that data sets containing polymorphs or going through a first-order phase transition are not included in the final structure set. Notably isosymmetric phase transitions without strong cell parameter discontinuities will get unnoticed by any automatic data mining procedure.

**CSD pressure check.** All structures with reported non-ambient pressure conditions were removed. However, the pressure flag is not always set which could lead to incorrect isobaric thermal expansion coefficients, such as for KARGIW, where the same authors measured the structure at room temperature at 0.2 kbar but also at 230 K at ambient pressure. The calculation of the thermal expansion coefficients on this compound led to a false negative volumetric expansion. Such cases can only be removed manually.

### Determination of thermal expansion coefficients

Expansivities were calculated using the Python NumPy library<sup>47</sup> following the method described by Cliffe & Goodwin.<sup>48</sup> A number of them were compared to the values obtained by using the on-line calculator PASCAL described in the Cliffe & Goodwin paper, but also with the unit strain calculations implemented in PLATON<sup>49</sup> and those on the Bilbao Crystallographic Server.<sup>50</sup> The linear expansion coefficients were thus calculated along three orthogonal directions defined by the eigenvectors of the unit strain matrix.

In the case of right angles between the cell parameters the uniaxial expansion coefficients coincide with those calculated along the crystallographic axes. When this is not the case, the

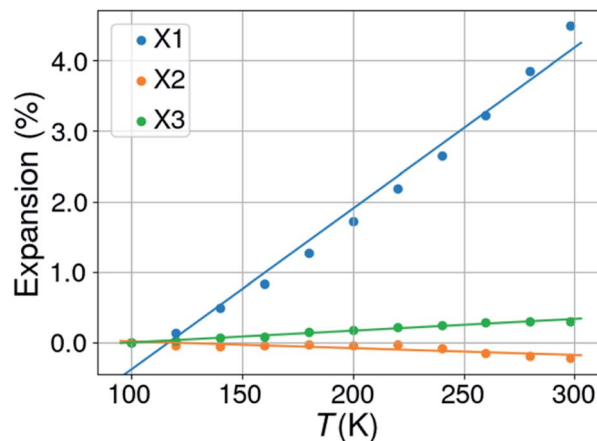


Fig. 2 Percentual thermal expansion along the three orthogonal directions in the structure of 3-ethyl-1-methylimidazolium nitrate (KUCPED).

coefficients are related to directions which are determined by the orthogonalization matrix of the cell metrics. With changing temperatures not only the cell axes change, but also the cell angles for triclinic and monoclinic space groups. In this case it can happen that all three cell axes increase in length with rising temperature, but that one or two orthogonal expansion coefficients are negative. An illustrating case is KUCPED<sup>51</sup> without any obvious negative uniaxial thermal expansion if only the cell axes are considered, but which has anyhow one slightly negative uniaxial thermal expansion, one close to zero expansion, and one normal positive expansion (Fig. 2 and Table 1).

For structures with more than 2 different temperatures, the mean uniaxial and volumetric expansion coefficients were determined by linear regression of the expansions  $E_i$  from the lowest ( $T_0$ ) to the highest temperature  $T_N$  with  $i = 0, 1 \dots N$ . This can be done in two different ways, either by calculating the expansion between subsequent points  $T_i$  and  $T_{i+1}$ :

$$E_j(T_i) = -1.0 + (1.0 + E_{i-1,j}) \times (1.0 + \alpha_{ij} \times (T_i - T_{i-1}))$$

Table 1 Cell parameters as a function of temperature in the structure of 3-ethyl-1-methylimidazolium nitrate<sup>a</sup>

$T$ (K)	$a$ (Å)	$b$ (Å)	$c$ (Å)	$\beta$ (°)
100	8.5306	15.778	12.4800	93.770
120	8.5418	15.7812	12.4741	93.738
140	8.5686	15.7871	12.4753	93.624
160	8.5954	15.7896	12.4797	93.519
180	8.6286	15.8000	12.4856	93.392
200	8.6642	15.8053	12.4859	93.265
220	8.6997	15.8123	12.4926	93.104
240	8.7341	15.8150	12.4898	92.942
260	8.7796	15.8229	12.4861	92.766
280	8.8267	15.8236	12.4876	92.544
298	8.8740	15.8256	12.4921	92.309

<sup>a</sup> Notes: CSD refcode KUCPED. Space group  $P2_1/n$ . Orthogonal thermal expansion coefficients:  $X_1 = 228.6$ ,  $X_2 = -9.7$ ,  $X_3 = 16.7 \text{ MK}^{-1}$ .



where  $\alpha_{ij}$  is eigenvalue  $j$  between temperatures  $T_{i-1}$  and  $T_i$  of the strain matrix. Alternatively, the expansion at each temperature can be calculated with respect to the lowest temperature.

$$E_j(T_i) = \alpha'_{ij} \times (T_i - T_0)$$

where  $\alpha'_{ij}$  is eigenvalue  $j$  of the unit strain matrix between temperature  $T_i$  and  $T_0$ . For  $N = 1$  the two methods are identical. The latter method is adopted by the algorithm in the PASCAL program and gives minor differences with the results when the expansion is calculated subsequently.

A problem may arise by the calculation of the eigenvalues and eigenvectors of the strain matrices at more than two temperatures. These are not always in the same order for subsequent calculations. The approach taken in PASCAL is to rearrange the eigenvalues and accompanying eigenvectors in ascending order, but this does not always give satisfactory results. The approach used here is to calculate the angle difference in the crystallographic system between the eigenvectors of the strain matrix between  $T_{i+1}$  and  $T_i$  and those between  $T_1$  and  $T_0$ . The minimum angle difference is taken to be the matching eigenvalue/eigenvector set, and eigenvalues and eigenvectors for the  $T_{i+1}/T_i$  strain matrix are eventually swapped to align them with those between  $T_1$  and  $T_0$ .

### Data sets used for thermal expansion properties analysis

From the 355 740 different organic entries in the CSD (5.41), a set of 37308 entries was extracted for 13840 different compounds (including polymorphics) with structures determined at two or more temperatures. After eliminating structures measured at non-ambient pressure, and those with abrupt cell parameter changes, unidentifiable temperatures, and those without space group information, a set of 7478 unique chemical compounds was extracted with data collected at at least two different temperatures and a minimum 50 K temperature interval between lowest and highest temperatures, yielding in total 20826 different entries (refcodes) and consequently an average of 2.78 different temperatures per unique compound. From the 20826 different refcodes, 580 entries were removed because of  $\langle U_{eq} \rangle_s$  values lying outside the trusted interval for their data collection temperature according to the criteria defined before. This could in addition lead to the removal of the chemical compound whenever only one refcode was left. The unit cells of all entries were transformed to the conventional unit cell where needed. Finally, 4719 unique structures comprising 4093 unique molecular compounds and 626 polymorphs have been retained for the analysis of the thermal expansion properties with 11658 different refcodes. The ESI† gives details for all 4719 structures: 14 157 ( $n(\alpha_l)$ ) axial and 4719 ( $n(\alpha_v)$ ) volumetric thermal expansion coefficients, 4719 Indicatix Anisotropy Coefficients (IAC, see below), number of entries per structure, data collection temperatures and hyperlinks to the original publications.

For the determination of the distributions of thermal expansion coefficients a further reduction was applied, since scatter in the data could seriously bias the obtained results. Subsets were defined consisting of only those values from

compounds with at least three data collections at different temperatures in a temperature interval larger than 50 K and for which a fit assuming a linear dependency on temperature yielded a goodness of fit value  $R^2$  of at least 0.75, 0.80, 0.85, and 0.90. As was outlined elsewhere<sup>48</sup> the dependency of the thermal expansion need not be linear, especially at temperatures well below the Debye temperature<sup>52</sup> or over continuous first-order (isosymmetric) phase transitions,<sup>53,54</sup> but in nearly all practical cases there is no need to go beyond a linear model.

These high-quality subsets contain proportionally more studies than the complete set of compounds in which the same compound was measured in the same laboratory on the same diffractometer and should be qualitatively more homogeneous than the complete set. In the data subset with coefficients determined from only 2 different temperature points, 23% originate from experiments performed in the same study, whereas for the subset with 3 different temperature points this is 26% (with 4: 36%, with 5: 42%, Fig. S1†).

## Results and discussion

### Distributions of thermal expansion coefficients

Fig. 3 shows the histograms of uniaxial and linear thermal expansion coefficients based on the data subset corresponding to linear fits with  $R^2 > 0.90$ , consisting of 745 volumetric and IAC values and 1129 axial values. The mean value ( $\mu$ ) of the volumetric thermal expansion of organic compounds is  $168.6 \text{ MK}^{-1}$  (median  $\tau = 168.3 \text{ MK}^{-1}$ ) with a standard deviation  $\sigma$  of  $72.5 \text{ MK}^{-1}$ . For the uniaxial expansion these numbers are  $\mu = 71.4$  ( $\tau = 62.6$ )  $\text{MK}^{-1}$  with  $\sigma = 69.9 \text{ MK}^{-1}$  and for the IAC  $-0.05$  ( $\tau = -0.05$ ) with  $\sigma = 0.41$ . The thermal coefficient samples are not normally distributed according to the D'Agostino and Pearson's test,<sup>55,56</sup> but the IACs are. The distribution of the axis thermal expansion coefficients is moderately skewed towards zero. The dependency of the distributions on  $R^2$  is relatively low (Table S1 and Fig. S2–S6†), and even the global data set containing 4719 observations has a distribution which is not completely different from the high-quality reduced data set. The obtained mean volumetric thermal and uniaxial expansion coefficients in this way are  $156.0$  and  $51.6 \text{ MK}^{-1}$ , respectively, with standard deviations of  $68.7$  and  $73.8 \text{ MK}^{-1}$  for 4719 different volumetric coefficients and 14 157 axial values. These mean values are somewhat smaller than those obtained using the reduced subsets and the standard deviations are somewhat larger for the volumetric coefficients, but smaller for the axial coefficients. The values from the global dataset are nevertheless relatively close to the values of the reduced subset, giving confidence that even cell parameters from studies in different laboratories performed on different machines can give valuable thermal expansion information.

It is interesting to note that the distribution of the IACs is very broad with a standard deviation of 0.41 between limiting values of  $-1$  and  $1$ , and it is slightly more symmetric than the distribution of the volumetric and uniaxial thermal expansion coefficient. The anisotropy of the thermal expansion could thus be considered as rare and large if  $-1 < \text{IAC} < -0.46$  or  $0.34 < \text{IAC} < 1.00$  and very rare and extraordinary large when  $-1 < \text{IAC} < -0.86$



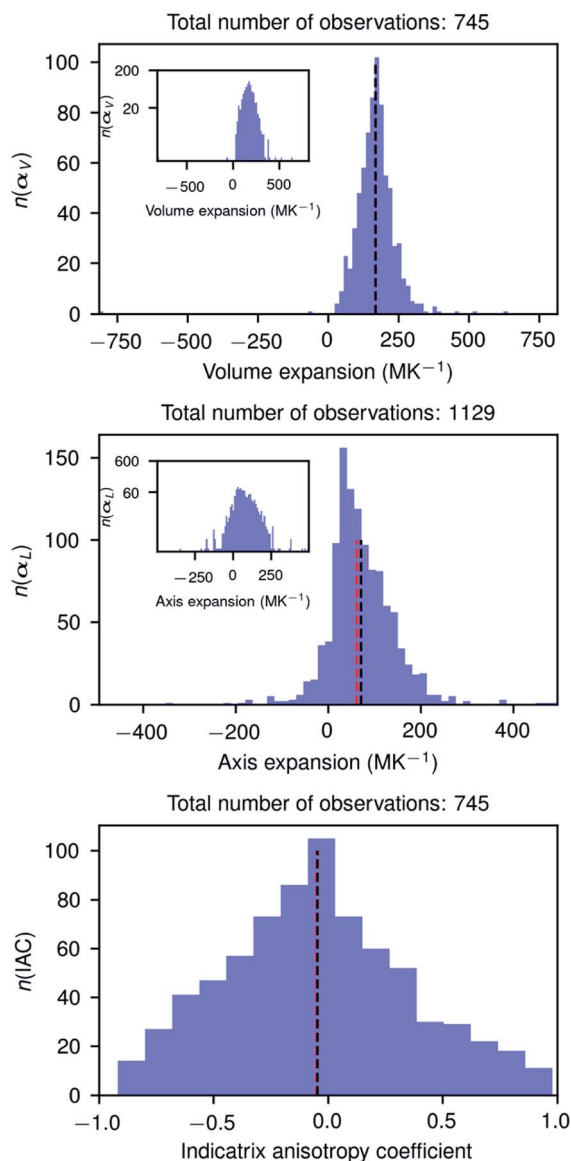


Fig. 3 Histograms of the volume and axis expansions and the indicatrix anisotropy coefficient of organic compounds in the CSD (version 5.41), calculated for compounds for which data were collected at least three different temperatures and for which the goodness of linear fit was at least 0.9. The optimal number of bins in each figure was determined using the Freedman–Diaconis rule (Freedman & Diaconis, 1981). The inset figures in the two upper plots show the same histogram but on a logarithmic scale. Black and red vertical dashed lines show the mean and median values, respectively, of the distributions. The numbers  $n$  on the vertical axes give the number of observations in each bin. The horizontal axes have been scaled to the minimum and maximum values of the distributions, respectively.

or  $0.74 < \text{IAC} < 1.00$ . In this light the extraordinary anisotropic thermal expansions recently reported for three photosalient crystals<sup>57</sup> with IACs of 0.248, 0.342, and  $-0.06$ , respectively, are rather normal. The three compounds are, however, metal–organic complexes whose structures could have a different IAC distribution. Although we have no formal proof of the IAC

distribution of metal–organic structures, we do not expect it to be very different from the IAC distribution of organic structures.

The highest positive volume expansion coefficient,  $2333.5 \text{ MK}^{-1}$ , is found for 3-benzoylpropionic acid whose structure was determined at two different temperatures by two different groups,<sup>58,59</sup> but needs further experimental verification using a strict experimental protocol (see also Fig. S6 in ESI†), preferably using data sets collected at more than 2 different temperatures spanning a large temperature interval.

## Two case studies

The methodology followed in this paper can be best illustrated by looking into more detail to the thermal expansion coefficients for compounds with a sufficiently large number of experimental structures determined at different temperatures.

**Glycine.** Glycine (CSD refcode GLYCIN) is a common amino acid which crystallizes in three different polymorphic forms at room temperature (Albrecht & Corey, 1939; Iitaka, 1958; Iitaka, 1960).<sup>60–62</sup> The  $\alpha$  and  $\beta$  forms crystallize in the monoclinic system with space groups  $P2_1/n$  and  $P2_1$ , respectively, whereas the  $\gamma$  form crystallizes in the enantiomorphic space group pair  $P3_1/P3_2$ . No systematic study of the thermal expansion coefficients of glycine has been reported, although Sun (2007) has reported on the thermal density evolution of glycine for the three polymorphs.<sup>36</sup> The CSD contains 100 different experimental structures for GLYCIN, of which two entries did not contain space group information and 42 structures were determined from pressure experiments. The remaining structures were divided into 4 different sets with space groups  $P2_1$  (9 refcodes),  $P2_1/n$  (35 refcodes),  $P3_1$  (4 refcodes) and  $P3_2$  (5 refcodes). The cell parameters of structures within the same 10 K temperature bin were averaged within each space group set and 5 refcodes were removed because of having  $\langle U_{\text{eq}} \rangle_s$  outside the ranges defined by the pink boxes in Fig. 1, leaving 3, 14, 2, and 3 unique refcodes for space groups  $P2_1$ ,  $P2_1/n$ ,  $P3_1$ , and  $P3_2$ , respectively, with 3, 6, 1, and 3 unique digital object identifiers (doi), respectively. Fig. 4 gives the thermal expansion plots for the different polymorphs of glycine and Table 2 presents the numeric data. Although the data are scarce for the  $\gamma$  polymorphs, it seems

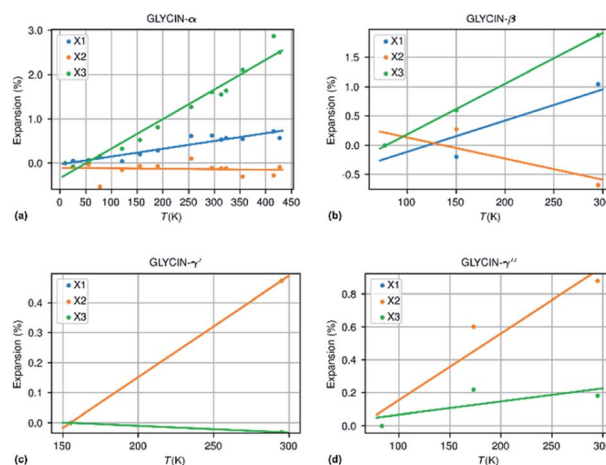


Fig. 4 Thermal expansion in the different polymorphs of glycine.

Table 2 Thermal expansion data of the polymorphs of glycine<sup>a</sup>

Compound	$N_m$	$\Delta T$ (K)	$\alpha_{X_1}$	$\alpha_{X_2}$	$\alpha_{X_3}$	$\alpha_V$	$\chi_\alpha$	IAC
Glycine- $\beta$	3	218	53.3	-36.1	86.4	103.6	2.3	0.06 (++)
Glycine- $\alpha$	14	417	17.6	-1.1	66.8	83.6	3.48	-0.20 (-++)
Glycine- $\gamma'$	2	140	33.7	33.7	-2.2	65.4	0.92	-0.93 (-++)
Glycine- $\gamma''$	3	212	40.4	40.4	8.0	89.3	1.89	-0.80 (+++)

<sup>a</sup> Notes: the thermal expansion coefficients  $\alpha$  are in  $\text{MK}^{-1}$ .  $\chi_\alpha$  is the NTE capacity and IAC the Indicatrix Anisotropy Coefficient.

that thermal expansion in that polymorph is smaller than that in the  $\alpha$  and  $\beta$  polymorphs. The statistics of the two different  $\gamma$  polymorphs are certainly too small to draw conclusions between thermal expansion differences between the different hands of polar structures. To the best of our knowledge, no systematic studies have been performed in this sense.

**Anthraquinone.** The second example concerns the thermal expansion of anthraquinone,  $\text{C}_{14}\text{H}_8\text{O}_2$  (refcode ANTQUO). Its structure has been reported 16 times in the CSD with a mixture of cell descriptions in  $P2_1/a$  (once with the  $a$ -axis as unique axis),  $P2_1/c$  and once  $P\bar{1}$ . The latter was discarded from further analysis and one structure (ANTQUO15) was removed from the analysis because of a strongly deviating cell parameter (doubled  $a$ -axis compared to the other structures). The cell parameters of six structures were averaged within the same 10 K temperature bin, leaving 8 structures for thermal expansion analysis. The results from 5 different studies have been included in this compilation. Fig. 5 shows the thermal expansion along the three orthogonal axes of anthraquinone; it gives an idea about the expected variability of the measurements performed at different times, on different laboratories, and on different diffractometers. The extracted thermal expansion coefficients are  $-35.6$ ,  $51.9$ ,  $94.8$  and  $110.9 \text{ MK}^{-1}$  for  $X_1$ ,  $X_2$ ,  $X_3$ , and  $V$ , respectively, with  $R^2$  goodness of fit parameters of  $0.52$ ,  $0.65$ ,  $0.93$ , and  $0.67$ , respectively.

### Negative volumetric expansion coefficients

Amazingly, 139 structures have been detected with negative volumetric thermal expansion. The different temperature data

for each of the compounds have been measured in different laboratories, and most probably on diffractometers of different brands. Additional tests were carried out on these entries to extract the most probable candidates for true negative thermal volumetric expansion. Table S2† gives the 127 compounds remaining after this selection with information about  $\alpha_V$ , the temperature interval over which it occurs, the number of temperature entries for each compound and the NTE capacity parameter  $\chi_\alpha$  which is the product of the relative volume contraction of the compound and the temperature range over which it is observed.<sup>63</sup> For each compound it was checked whether  $\langle U_{\text{eq}} \rangle_s$  showed a positive tendency with temperature. If not, it was labeled 's'. The structures labeled 'o' have been reported in the same publication – but not necessarily under the same conditions. For a number of structures, the reliability was checked by inspecting the original publications.

Two compounds with extremely large negative volume expansion coefficients ( $-814.36 \text{ MK}^{-1}$ , UFERED;<sup>64</sup>  $-653.79 \text{ MK}^{-1}$ , KARGIW<sup>65</sup>) were eliminated from this set since the refcodes for each compound contained a mixture of structures measured at ambient and not ambient pressures and different temperatures. The non-ambient pressure structures had not been flagged as such in the CSD entries.

The crystals of other structures appeared to have different colors at different temperatures. Assignment of color is rather subjective. Although this could indicate a phase transition, it could also be due to a wrong assignment of the color, due to the subjectivity of the assignment process or even due to the physical inability of the researcher to differentiate specific colors.

The histogram of the collected uniaxial thermal expansion coefficients shows that nearly 40% of the values are negative. Out of the 4719 unique compounds 33.1% have one negative thermal coefficient, 5.3% two, and 1.5% three. Only 60.0% of the compounds have three strictly positive uniaxial thermal coefficients, which contradicts the general belief that negative thermal expansion in organic compounds is very rare.<sup>19,63,66,67</sup> Isotropic or volumetric negative thermal expansion is however indeed very rare in organic compounds, with only a few confirmed cases.

### Anisotropy of thermal expansion

Thermal expansion is rarely completely isotropic – except for compounds crystallizing in cubic space groups – but usually anisotropic to very anisotropic. In order to quantify the anisotropy, the indicatrix anisotropy coefficient was calculated following the method proposed by Doube for quantifying rods, plates, and intermediate forms in 3D ellipsoidal geometries.<sup>68</sup> When  $X_1$ ,  $X_2$ , and  $X_3$  are the lengths of the principal axes of an ellipsoid, and  $X_1 \leq X_2 \leq X_3$ , then  $\text{IAC} = X_1/X_2 - X_2/X_3$ .

The Indicatrix Anisotropy Coefficient (IAC) ranges from  $-1$  for very oblate ellipsoids to  $+1$  for strongly prolate ellipsoids when the lengths of the principal axes are all positive. However, in the case of the thermal expansion indicatrix axes may have negative lengths, which means that IAC may range from  $-\infty$  to  $+\infty$ . A slightly modified definition is therefore

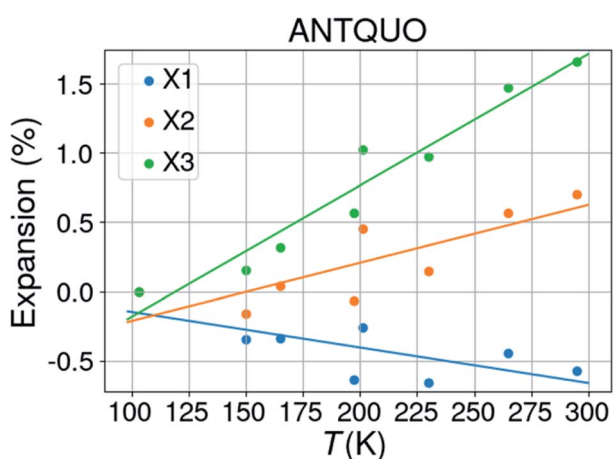


Fig. 5 Thermal expansion of anthraquinone.



used here by using the absolute values of  $X_1$ ,  $X_2$ , and  $X_3$  and adding a suffix to IAC indicating the sign of each value. Thus for example  $IAC = 0.231(++-)$  means that the value of IAC is 0.231 and that  $X_1$ ,  $X_2$  have a positive sign and  $X_3$  a negative sign, but also that the absolute value of  $X_3$  is larger than that of  $X_1$  and  $X_2$ . With one or more negative thermal expansion coefficients the ellipsoids are no longer oblate or prolate, but the notion of strong anisotropy remains when IAC is close to 1 or  $-1$ .

Fig. 3 shows an approximately symmetrical distribution of IAC values, with mean and median values slightly lower than 0.00. A significant proportion of the compounds displays extreme anisotropic thermal expansion of which four examples are given in Fig. 6. They have either very prolate (ACEMID and RAWBEW) or very oblate (TBPHAN & FIFMAI) IAC's. The structures with prolate IAC's have thermal expansions which are close to zero along two principal axes, and in these two cases also along two crystallographic axes ( $a$  and  $b$  for ACEMID and  $b$  and  $c$  for RAWBEW, respectively), and a strongly positive thermal expansion along one axis ( $c$  for ACEMID and  $a$  for RAWBEW). The structures with strongly oblate IAC's have only one axis with close to zero thermal expansion ( $c$  for TBPHAN and  $a$  for FIFMAI) and two axes with strong positive thermal expansion ( $a$  and  $b$  for TBPHAN and  $b$  and  $c$  for FIFMAI).

From the crystal packing it is not always clear why the thermal expansion is very strong in one or two directions and close to zero in the other directions. In the case of the acetamide structure the hydrogen bond network is clearly three dimensional without any preferential directions. The biguanidinium bis(dinitramide) structure does have a quasi two-dimensional structure with the hydrogen network parallel to the  $bc$  plane and thus only very weak van der Waals type interactions between the biguanidinium bis(dinitramide). Although this gives a qualitative explanation why the expansion in the direction perpendicular to the layers can be much larger than the expansion within the layers, it does not give a quantitative explanation why the latter is so close to zero. For the structures

with extreme oblate IAC's the packing does not give clear clues why the expansion is strongly positive in two directions and close to zero in the third direction. TBPHAN does not have any classical hydrogen bonds, but rather halogen bonds which are not directed along any of the crystallographic axes. In the structure of FIFMAI there is one infinite hydrogen bond chain running along the  $c$ -axis, and only weak Waals interactions along the other two directions. The *ab initio* calculation of thermal expansion coefficients for molecular materials in a quasi-harmonic approximation or using full molecular dynamics simulations is still in its infancy although some progress is made.<sup>69–72</sup> The crucial parameters of these calculations are related to the correct description of a wide variety of interactions present in molecular compounds, such as covalent and electrostatic interactions or hydrogen bond and long-range dispersive interactions. This makes it difficult to predict *ab initio* the thermal expansion properties of even very simple but highly anisotropic systems such as graphite and boron nitride.

### Organic negative thermal expansion compounds

The compounds crystallizing in the structures presented in Table S2† which have possible negative volumetric expansion belong to a wide range of classes of compounds of varying complexity, with no apparent clear trend. Also, from a crystallographic point of view, there seems to be no particular clustering of the structures in specific space groups. This study highlighted very simple molecules, such as bis(ammonium) carbonate monohydrate  $((\text{NH}_4)_2\text{CO}_3 \cdot \text{H}_2\text{O}; \text{KOYRUN})$  or carbazole  $(\text{C}_{12}\text{H}_9\text{N}; \text{CRBZOL})$ , but also chiral  $Z' = 4$  (Ibuprofen) $2(4,4'$ -dipyridyl) co-crystals (IJIAN) or large  $\epsilon$ -cyclodextrin hydrate (NOBBOV) structures. While some of the structures with the lowest thermal expansion coefficient belong to compounds which are hydrogen rich and contain long aliphatic chains, there are also examples of structures containing no hydrogen atoms at all, such as the fluorinated fullerene derivatives WUXNIM and WEDMOI. In the same manner, while most structures are neutral compounds, there are several examples of salts. Furthermore, no heavy atom effect was evident, with iodine being the heaviest element present in the list.

As expected, this high structural variability of the compounds reflects in the wide range and type of intermolecular forces binding the structures together. All types of intermolecular interactions expected in organic compounds were observed in the list, ranging from strong ionic or charge-assisted hydrogen bonds, to weak  $\pi$ -stacking,  $\text{CH} \cdots \pi$  interactions and dihydrogen contacts.

The inability to draw any qualitative structure–property relationship for negative thermal expansion can be attributed on the one hand to the complexity of the balance between the different types of intermolecular interactions and molecular flexibility contributing to the thermal expansion, and on the other hand to the paucity of literature data on any series of compounds. To our knowledge there are no structural studies dealing with the thermal expansion of a homologous series of organic compounds or their salts. This is mainly due to the fact that, in spite of the increasing availability of diffraction as an

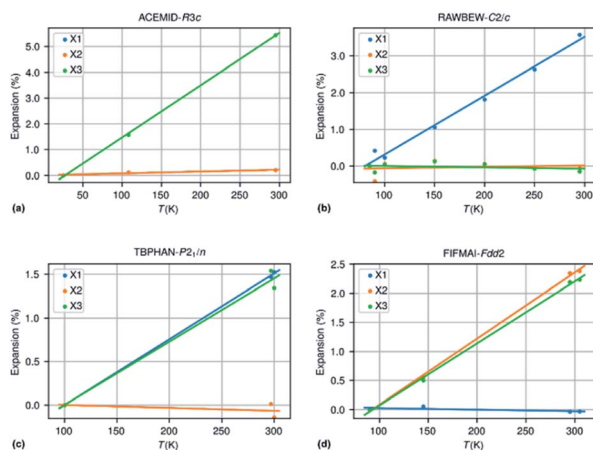


Fig. 6 (a) ACEMID – acetamide; IAC = 0.96 (three different studies); (b) RAWBEW – biguanidinium bis(dinitramide); IAC = 0.94 (three different studies); (c) TBPHAN – tetrabromophthalic anhydride; IAC =  $-0.91$ ; (2 different studies); (d) FIFMAI – 2-biphenylol IAC =  $-0.91$  (1 study).



analytical tool, one X-ray structure is still generally considered adequately sufficient for characterizing an entire series in most synthetic studies.

### Experimental validation of negative thermal expansion compounds

The list of structures presented in the current work does not give more than a couple of hints where to look for zero or negative thermal expansion properties. Experimental validation needs to be carried out in the same laboratory on the same diffractometer using a strict thermal and experimental protocol and preferably on the same crystal, although one could think that in order to have better statistics multiple observations of the same structure of a chemical compound should be used in order to obtain a more precise estimate of the thermal expansion coefficients. It could be argued that powder diffraction would be in this sense actually the preferred method for thermal expansion experiments, since the experiment averages over all crystallites present in the sample.<sup>73</sup> However, high accuracy in powder diffraction can also only be obtained by strict experimental protocols and proper alignment of the diffractometer. Comparing cell parameters for the same crystalline phase determined on different powder diffractometers give in general different results. Powder diffraction could therefore give higher precision than single crystal diffraction, but not necessarily higher accuracy ("precise estimates are not necessarily accurate").<sup>74–77</sup> A number of single-crystal studies also use powder diffraction analyses, which are much more finely sampled in temperature. A close inspection of several combined powder/single crystal studies gives rather important differences between cell parameters determined by powder diffraction on the one hand and single-crystal diffraction on the other hand. An example is  $\text{Mg}(\text{BH}_4)_2$  (refcode BOWMIK<sup>78</sup>) – which is flagged as 'organic' in the CSD – where the single-crystal study performed with synchrotron radiation at 100 K yields a cell volume of 3440 Å<sup>3</sup>, but the powder diffraction study, also performed with synchrotron radiation, gave 3425 Å<sup>3</sup> at 100 K and 3431 Å<sup>3</sup> at room temperature. The CSD only reports the single crystal study at 100 K (BOWMIK) and the powder diffraction study at 298 K (BOWMIK01), suggesting therefore that the compound displays negative volumetric thermal expansion, which is clearly not the case as evidenced by the finely sampled cell parameter values determined by powder diffraction between 100 and 500 K. This emphasizes the need to collect data at more than two data points for the extraction of thermal expansion coefficients, since scatter in the data may lead to less reliable values.

Two major sources of bias are introduced in the results of this data mining study. The first bias is of course connected to the accuracy of the reported cell parameters. It is not certain that cell parameters from recent studies are necessarily more accurate than cell parameters from studies reported 2–4 decades ago. Related to this is the accurate determination of the temperature at the position of the crystal and its stability during the measurement. The cif-files of more recent structure determination contain often a standard uncertainty for the data

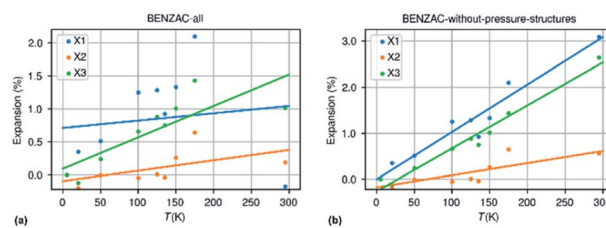


Fig. 7 Thermal expansion data for benzoic acid extracted from the CSD with pressure data incorrectly included (a) and with only data collected at ambient pressure but different temperatures (b).

collection temperature, since the temperature can be logged continuously during the measurement, but this gives only a hint about the stability of the temperature, not on its accuracy.

The second bias is related to inaccurate or missing information in the cif-files that are deposited at the CSD. A major source of error is the presence of structures measured at non-ambient pressure, usually at room temperature but not only, and which are not flagged as such in the cif-file. The structure of benzoic acid is reported 21 times in the CSD of which 11 have been recorded at room temperature. None are flagged as structures determined at non-ambient pressure. Out of the 11 room temperature structures 7 appeared to originate from the same study by Caia & Katrusiak.<sup>79</sup> Fig. 7 shows the expansion curves of benzoic acid with and without the structures from the Caia & Katrusiak study.

### Normal and anomalous values of thermal expansion coefficients

The present study gives an estimate of the mean thermal expansion coefficient to be expected for the crystalline structures of organic compounds. It shows as well that the dispersion of expected values is relatively large. With the actual values in hand it is easier to define when a reported thermal expansion coefficient is anomalous or exceptional. Since the thermal expansion coefficients are far from normally distributed, the Bienaymé–Chebyshev inequality should be applied stating that a minimum of 75% of values must lie within two standard deviations of the mean and 89% within three standard deviations. If a threshold of three standard deviations is retained for the volumetric thermal expansion it means that whenever a reported value is lower than  $-48.7 \text{ MK}^{-1}$  or larger than  $386.3 \text{ MK}^{-1}$ , it may be considered as anomalous. For an axial thermal expansion coefficient these values are  $-138.3$  and  $281.1 \text{ MK}^{-1}$ , respectively.

## Conclusions and outlook

Data mining studies rely on the availability, diversity and redundancy of data. The Cambridge Structural Database is an extremely important source of available data from very different compounds, and from many different sources and at different times. The importance of redundant data inside this huge data base cannot be underestimated, since this helps in the worst case to discard faulty data and in the best case to increase



precision. Therefore, scientists should be even more encouraged than they are now to deposit data from structures that have been published before. All practical crystallographers make very often the decision to skip the full data collection once they discover from the preliminary cell screening and 'helped' by fast reduced cell checkers as CellCheckCSD<sup>80</sup> that can be enabled during data collection, that the structure has already been published. In the best case they do not discover this or they decide to perform the data collection anyhow. Then they should take the time to refine the structure and to complete the cif-file, and to deposit them at the CSD labeled as a private communication and, more generally, as a gift to science. This is the best case scenario to advance science from data mining projects.

The main conclusion of this study is that negative uniaxial thermal expansion in organic crystal structures is much more common than generally believed. Most papers dealing with negative thermal expansion state that this is a very rare phenomenon, especially in organic crystal structures. We show that about a third of the organic crystal structures may present negative uniaxial thermal expansion. We show – also against general belief – that thermal expansion is commonly very anisotropic in organic crystal structures. We identify about 100 crystal structures which may present negative volumetric thermal expansion. The data present in the Cambridge Structural Database are not 100% trustworthy, which could bias the results of data mining studies as the current one. We present a new simple method to validate or invalidate temperature data in the CSD by an internal check based on the average equivalent atomic displacement parameter of the organic structure is proposed to validate or invalidate the reported temperature in the CSD entry.

## Author contributions

The conceptualization, investigation, and validation of the methods and results presented in this study were done by both authors. AvdL performed the methodology and data curation stages and developed the Python scripts used in this study. The paper was written jointly.

## Conflicts of interest

There are no conflicts to declare.

## References

- 1 J. D. James, J. A. Spittle, S. G. R. Brown and R. W. Evans, *Meas. Sci. Technol.*, 2001, **12**, R1–R15.
- 2 K. Takenaka, Y. Okamoto, T. Shinoda, N. Katayama and Y. Sakai, *Nat. Commun.*, 2017, **8**, 14102.
- 3 J. Mohanraj, E. Capria, L. Benevoli, A. Perucchi, N. Demitri and A. Fraleoni-Morgera, *Phys. Chem. Chem. Phys.*, 2018, **20**, 1984–1992.
- 4 Y. N. Heit and G. J. O. Beran, *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.*, 2016, **72**, 514–529.
- 5 L. Glasser and H. D. B. Jenkins, *J. Chem. Eng. Data*, 2011, **56**, 874–880.
- 6 L. Glasser, *J. Phys. Chem. Solids*, 2012, **73**, 139–141.
- 7 K. Takenaka and H. Takagi, *Appl. Phys. Lett.*, 2009, **94**, 131904.
- 8 A. E. Phillips, G. J. Halder, K. W. Chapman, A. L. Goodwin and C. J. Kepert, *J. Am. Chem. Soc.*, 2010, **132**, 10–11.
- 9 X. Song, Z. Sun, Q. Huang, M. Rettenmayr, X. Liu, M. Seyring, G. Li, G. Rao and F. Yin, *Adv. Mater.*, 2011, **23**, 4690–4694.
- 10 X. Jiang, M. S. Molokeev, P. Gong, Y. Yang, W. Wang, S. Wang, S. Wu, Y. Wang, R. Huang, L. Li, Y. Wu, X. Xing and Z. Lin, *Adv. Mater.*, 2016, **28**, 7936–7940.
- 11 J. Chen, Q. Gao, A. Sanson, X. Jiang, Q. Huang, A. Carnera, C. G. Rodriguez, L. Olivi, L. Wang, L. Hu, K. Lin, Y. Ren, Z. Lin, C. Wang, L. Gu, J. Deng, J. P. Attfield and X. Xing, *Nat. Commun.*, 2017, **8**, 14441.
- 12 P. Lightfoot, D. A. Woodcock, M. J. Maple, L. A. Villaescusa and P. A. Wright, *J. Mater. Chem.*, 2001, **11**, 212–216.
- 13 D. Dubbeldam, K. S. Walton, D. E. Ellis and R. Q. Snurr, *Angew. Chem., Int. Ed.*, 2007, **46**, 4496–4499.
- 14 S. S. Han and W. A. Goddard, *J. Phys. Chem. C*, 2007, **111**, 15185–15191.
- 15 A. F. Sapnik, H. S. Geddes, E. M. Reynolds, H. H. M. Yeung and A. L. Goodwin, *Chem. Commun.*, 2018, **54**, 9651–9654.
- 16 F.-X. Coudert and J. D. Evans, *Coord. Chem. Rev.*, 2019, **388**, 48–62.
- 17 A. Sanson and J. Chen, *Front. Chem.*, 2019, **7**, 284.
- 18 A. Sanson, *Mater. Res. Lett.*, 2019, **7**, 412–417.
- 19 K. Takenaka, *Sci. Technol. Adv. Mater.*, 2012, **13**, 013001.
- 20 M. T. Dove and H. Fang, *Rep. Prog. Phys.*, 2016, **79**, 066503.
- 21 K. W. Chapman and P. J. Chupas, *J. Am. Chem. Soc.*, 2007, **129**, 10090–10091.
- 22 C. R. Morelock, B. K. Greve, L. C. Gallington, K. W. Chapman and A. P. Wilkinson, *J. Appl. Phys.*, 2013, **114**, 213501.
- 23 J. Zhu, J. Zhang, H. Xu, S. C. Vogel, C. Jin, J. Frantti and Y. Zhao, *Sci. Rep.*, 2014, **4**, 3700.
- 24 L. C. Gallington, B. R. Hester, B. S. Kaplan and A. P. Wilkinson, *J. Solid State Chem.*, 2017, **249**, 46–50.
- 25 Y. Grosu, A. Faik, J.-M. Nedelec and J.-P. Grolier, *J. Phys. Chem. C*, 2017, **121**, 11499–11507.
- 26 W. L. Queen, C. M. Brown, D. K. Britt, P. Zajdel, M. R. Hudson and O. M. Yaghi, *J. Phys. Chem. C*, 2011, **115**, 24915–24919.
- 27 M. Santoro, F. A. Gorelli, R. Bini, J. Haines and A. van der Lee, *Nat. Commun.*, 2013, **4**, 1557.
- 28 N. C. Burtch, S. J. Baxter, J. Heinen, A. Bird, A. Schneemann, D. Dubbeldam and A. P. Wilkinson, *Adv. Funct. Mater.*, 2019, **29**, 1904669.
- 29 J. Chen, L. Hu, J. Deng and X. Xing, *Chem. Soc. Rev.*, 2015, **44**, 3522–3567.
- 30 J. N. Grima, P. S. Farrugia, R. Gatt and V. Zammit, *Proc. R. Soc. A*, 2007, **463**, 1585–1596.
- 31 E. R. Engel, V. J. Smith, C. X. Bezuidenhout and L. J. Barbour, *Chem. Mater.*, 2016, **28**, 5073–5079.
- 32 L. Zhao and C. Zhong, *J. Phys. Chem. C*, 2009, **113**, 16860–16862.
- 33 M. Szafranski, *J. Mater. Chem. C*, 2013, **1**, 7904–7913.



- 34 H. Liu, M. J. Gutmann, H. T. Stokes, B. J. Campbell, I. R. Evans and J. S. O. Evans, *Chem. Mater.*, 2019, **31**, 4514–4523.
- 35 C. R. Groom, I. J. Bruno, M. P. Lightfoot and S. C. Ward, *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.*, 2016, **72**, 171–179.
- 36 C. C. Sun, *J. Pharm. Sci.*, 2007, **96**, 1043–1052.
- 37 P. Sanschagrin, *Acta Crystallogr., Sect. A: Found. Adv.*, 2017, **73**, a67.
- 38 E. M. Melchor-Martínez, D. A. Silva-Mares, E. Torres-López, N. Waksman-Minsky, G. F. Pauli, S.-N. Chen, M. Niemitz, M. Sánchez-Castellanos, A. Toscano, G. Cuevas and V. M. Rivas-Galindo, *J. Nat. Prod.*, 2017, **80**, 2252–2262.
- 39 C. F. Macrae, I. Sovago, S. J. Cottrell, P. T. A. Galek, P. McCabe, E. Pidcock, M. Platings, G. P. Shields, J. S. Stevens, M. Towler and P. A. Wood, *J. Appl. Crystallogr.*, 2020, **53**, 226–235.
- 40 G. I. Nikonov, L. G. Kuzmina, D. A. Lemenovskii and V. V. Kotov, *J. Am. Chem. Soc.*, 1996, **118**, 6333.
- 41 G. I. Nikonov, L. G. Kuzmina, D. A. Lemenovskii and V. V. Kotov, *J. Am. Chem. Soc.*, 1995, **117**, 10133–10134.
- 42 F. N. David and J. W. Tukey, *Biometrics*, 1977, **33**, 768.
- 43 M. Nespolo and M. I. Aroyo, *Acta Crystallogr., Sect. A: Found. Adv.*, 2016, **72**, 523–538.
- 44 J. Rodríguez-Carvajal and J. González-Platas, *Acta Crystallogr., Sect. A: Found. Crystallogr.*, 2005, **61**, c22.
- 45 T. Roisnel, CRYSCALC, Centre de Diffractométrie X, UMR6226 CNRS Université de Rennes I, Institut des Sciences Chimiques de Rennes, 35042 Rennes Cedex, France, 2020.
- 46 L. C. Andrews, H. J. Bernstein and G. A. Pelletier, *Acta Crystallogr., Sect. A: Found. Crystallogr.*, 1980, **36**, 248–252.
- 47 C. R. Harris, K. J. Millman, S. J. van der Walt, R. Gommers, P. Virtanen, D. Cournapeau, E. Wieser, J. Taylor, S. Berg, N. J. Smith, R. Kern, M. Picus, S. Hoyer, M. H. van Kerkwijk, M. Brett, A. Haldane, J. F. del Río, M. Wiebe, P. Peterson, P. Gérard-Marchant, K. Sheppard, T. Reddy, W. Weckesser, H. Abbasi, C. Gohlke and T. E. Oliphant, *Nature*, 2020, **585**, 357–362.
- 48 M. J. Cliffe and A. L. Goodwin, *J. Appl. Crystallogr.*, 2012, **45**, 1321–1329.
- 49 A. L. Spek, *Acta Crystallogr., Sect. D: Biol. Crystallogr.*, 2009, **65**, 148–155.
- 50 E. S. Tasci, G. de la Flor, D. Orobengoa, C. Capillas, J. M. Perez-Mato and M. I. Aroyo, *EPJ Web Conf.*, 2012, **22**, 00009.
- 51 J. S. Wilkes and M. J. Zaworotko, *J. Chem. Soc., Chem. Commun.*, 1992, 965.
- 52 W. I. F. David, R. M. Ibberson and T. Matsuo, *Proc. R. Soc. London, Ser. A*, 1993, **442**, 129–146.
- 53 A. van der Lee, G. H. Roche, G. Wantz, J. J. E. Moreau, O. J. Dautel and J.-S. Filhol, *Chem. Sci.*, 2018, **9**, 3948–3956.
- 54 D. G. Dumitrescu, G. H. Roche, J. J. E. Moreau, O. J. Dautel and A. van der Lee, *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.*, 2020, **76**, 661–673.
- 55 R. B. D'Agostino, *Biometrika*, 1971, **58**, 341–348.
- 56 R. D'Agostino and E. S. Pearson, *Biometrika*, 1973, **60**, 613–622.
- 57 K. Yadava, G. Gallo, S. Bette, C. E. Mulijanto, D. P. Karothu, I.-H. Park, R. Medishetty, P. Naumov, R. E. Dinnebier and J. J. Vittal, *IUCrJ*, 2020, **7**, 83–89.
- 58 S. Selladarai, M. S. Kumar and K. Subramanian, *Proc. Indian Acad. Sci., Chem. Sci.*, 1990, **102**, 39–43.
- 59 H. W. Thompson, P. A. Vanderhoff and R. A. Lalancette, *Acta Crystallogr., Sect. C: Cryst. Struct. Commun.*, 1991, **47**, 1443–1445.
- 60 G. Albrecht and R. B. Corey, *J. Am. Chem. Soc.*, 1939, **61**, 1087–1103.
- 61 Y. Iitaka, *Acta Crystallogr.*, 1958, **11**, 225–226.
- 62 Y. Iitaka, *Acta Crystallogr.*, 1960, **13**, 35–45.
- 63 C. S. Coates and A. L. Goodwin, *Mater. Horiz.*, 2019, **6**, 211–218.
- 64 L. E. Connor, A. D. Vassileiou, G. W. Halbert, B. F. Johnston and I. D. H. Oswald, *CrystEngComm*, 2019, **21**, 4465–4472.
- 65 R. Lee, D. S. Yufit, M. R. Probert and J. W. Steed, *Cryst. Growth Des.*, 2017, **17**, 1647–1653.
- 66 C. Lind, *Materials*, 2012, **5**, 1125–1154.
- 67 J. Grima, V. Zammit and R. Gatt, *Xjenza*, 2006, **11**, 17–29.
- 68 M. Doube, *Front. Endocrinol.*, 2015, **6**, 15.
- 69 A. Erba, J. Maul and B. Civalieri, *Chem. Commun.*, 2016, **52**, 1820–1823.
- 70 J. G. Brandenburg, J. Potticary, H. A. Sparkes, S. L. Price and S. R. Hall, *J. Phys. Chem. Lett.*, 2017, **8**, 4319–4324.
- 71 H.-Y. Ko, R. A. DiStasio, B. Santra and R. Car, *Phys. Rev. Mater.*, 2018, **2**(5), 055603.
- 72 J. Hoja, H.-Y. Ko, M. A. Neumann, R. Car, R. A. DiStasio and A. Tkatchenko, *Sci. Adv.*, 2019, **5**, eaau3338.
- 73 J. I. Langford and D. Louër, *Rep. Prog. Phys.*, 1996, **59**, 131–234.
- 74 D. Schwarzenbach, S. C. Abrahams, H. D. Flack, W. Gonschorek, T. Hahn, K. Huml, R. E. Marsh, E. Prince, B. E. Robertson, J. S. Rollett and A. J. C. Wilson, *Acta Crystallogr., Sect. A: Found. Crystallogr.*, 1989, **45**, 63–75.
- 75 W. Parrish, *Acta Crystallogr.*, 1960, **13**, 838–850.
- 76 J. A. Kaduk, *Adv. X-Ray Anal.*, 1997, **40**, 352–370.
- 77 F. H. Herbstein, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2000, **56**, 547–557.
- 78 Y. Filinchuk, R. Černý and H. Hagemann, *Chem. Mater.*, 2009, **21**, 925–933.
- 79 W. Cai and A. Katrusiak, *CrystEngComm*, 2012, **14**, 4420.
- 80 P. Wood, CCDC Newsletter, November 2011.

