Supertetrahedral polyanionic network in the first lithium phosphidoindate Li₅InP₂ – structural similarity to Li₂SiP₂ and Li₂GeP₂ and dissimilarity to Li₃AlP₂ and Li₃GaP₂†

Tassilo M. F. Restle,a Volker L. Deringer,ab Jan Meyer,a Gabriele Raudaschl-Siebera and Thomas F. Fässlerb *a

Phosphide-based materials have been investigated as promising candidates for solid electrolytes, among which the recently reported Li₉AlP₄ displays an ionic conductivity of 3 mS cm⁻¹. While the phases Li–Al–P and Li–Ga–P have already been investigated, no ternary indium-based phosphide has been reported up to now. Here, we describe the synthesis and characterization of the first lithium phosphidoindate Li₅InP₂, which is easily accessible via ball milling of the elements and subsequent annealing. Li₅InP₂ crystallizes in the tetragonal space group I4/acd with lattice parameters of a = 12.0007(2) and c = 23.917(5) Å, featuring a supertetrahedral polyanionic framework of interconnected InP₄ tetrahedra. All lithium atoms occupy tetrahedral voids with no partial occupation. Remarkably, Li₅InP₂ is not isotypic to the previously reported homologues Li₃AlP₂ and Li₃GaP₂, which both crystallize in the space group Cmce and feature 2D layers of connected tetrahedra but no supertetrahedral framework. DFT computations support the observed stability of Li₅InP₂. A detailed geometrical analysis leads to a more general insight into the structural factors governing lithium ion mobility in phosphide-based materials: in the non-ionic conducting Li₃InP₂ the Li ions exclusively occupy tetrahedral voids in the distorted close packing of P atoms, whereas partially filled octahedral voids are present in the moderate ionic conductors Li₂SiP₂ and Li₂GeP₂.

Furthermore, we extended the system to the heavier tetrel (group-14) homologues, phosphidogermanimates, with two Li-ion conducting modifications of Li₅GeP₄ that show ionic conductivities of up to 8.6 × 10⁻⁵ S cm⁻¹ and with Li₁₈GeP₄₆, which achieves an ionic conductivity of 1.7 × 10⁻³ S cm⁻¹.¹⁰ The structural building units in these phosphides are [TP₄]⁻₈⁻tetrahedra surrounded by lithium atoms (T = Si, Ge). They exhibit a huge structural variety, and by decreasing the amount of lithium, condensed and covalently connected tetrahedra are formed, thereby offering different polyanionic networks: Li₁₀Si₂P₆ features pairs of edge-sharing SiP₄ tetrahedra,¹² in Li₅SiP₂/Li₂GeP₂ and Li₁₀Si₄P₆, respectively, SiP₄ and GeP₄ tetrahedra are condensed to networks of supertetrahedra.¹³,¹⁴ Layered structures have been reported as well: in Li₁₃Si₃P₁₀, vertex-sharing SiP₄ tetrahedra form double layers,¹₂ and Li₆Ge₃P₁₃ is built up by a two dimensionally extended polyanion comprising GeP₄ and Ge(P₃Ge) tetrahedra.¹³

Phosphide-based materials as lithium ionic conductors originated from the aliovalent substitution of [TtS₄]⁻₈⁻tetrahedra, which are the main building block in sulfide-based conductors. This leads to analogous structures with more negatively charged [TP₄]⁻₈⁻tetrahedra, which can therefore accommodate more lithium than the well-known sulfur-based...
analogues. In recent investigations we expanded this class of compound further to phosphidoaluminates, which contain tetrahedral AlP4 building units, and we discovered the fast lithium ion conductor Li3AlP2, which shows ionic conductivities of $3 \times 10^{-3}$ S cm$^{-1}$. In addition, we also obtained Li3AlP2, which is built up by $2/3[AlP_4^{3-}]$ layers of corner- and edge-sharing AlP4 tetrahedra, and we then also introduced the isotypic gallium compound Li3GaP2, as the first phosphidogallate. Both triolate (Tr = Al, Ga) compounds do not show moderate lithium ion conductivity but unexpectedly turned out to be direct band gap semiconductors with optical band gaps of 3.1 and 2.8 eV, respectively.

Prior to the present work, no ternary Li–In–P phase has been described in the literature, and only one ternary Na–In–P phase was mentioned: Na3InP2 is built up by a distorted hcp of P atoms with all octahedral voids filled by Na, whereas the tetrahedral voids are occupied by Na and In, giving rise to a poly-anionic network of corner-sharing InP4 tetrahedra.

In the present work, we report the first lithium phosphidoindate, Li3InP2, synthesized via ball milling of the elements and subsequent annealing. The compound retains the principal structural building unit of TrP4 tetrahedra, but their arrangement is notably different from that of Li3AlP2 and Li3GaP2. In Li3InP2, the InP4 tetrahedra are condensed to supertetrahedra in a three-dimensional framework. The structure is determined by single crystal X-ray diffraction and analyzed by complementary solid-state NMR experiments and first-principles computations. The knowledge of the lithium ion mobilities of the now-completed series of phosphidoalaterates allows us to suggest a structural design rule linked to ionic conductivity, namely, the presence (or absence) of partially occupied Li-containing octahedral sites between which the ions can move rapidly.

**Results and discussion**

**Synthesis and structure of Li3InP2**

Li3InP2 was synthesized from the elements via a two-step procedure. At first, stoichiometric amounts of lithium, indium and phosphorus were ball milled resulting in a reactive mixture. Besides small amounts of the desired phase, Li3InP2, the polycrystalline powder contains considerable amounts of InP and Li10In17, as impurities (see Fig. S4†). Subsequently, pellets of the mixture were annealed in sealed niobium ampules at 1023 K for 22 h. Afterwards, the ampoules were rapidly cooled to room temperature by quenching in an ice-water mixture yielding almost phase-pure Li3InP2 with 3.3(1)% Li10In17 as an impurity according to Rietveld analysis (Fig. S3†). Annealing at lower temperatures such as 673 K or slow cooling rates led to impurities such as InP. Powdered Li3InP2 is brick-red. Complete data of the Rietveld refinement are given in the ESI; Tables S5 and S6.†

Red single crystals of Li3InP2 were obtained after reacting the elements with the formal stoichiometry “Li10In17P3” at 1073 K in tantalum ampules. Besides Li3InP2, the resulting product contains InP and at least one more, so far unknown phase according to unassigned reflections in the powder X-ray diffractogram (see Fig. S5†). Details of the structure refinement of the single crystal X-ray diffraction data of Li3InP2 are listed in the ESI in Tables S1–S4.†

According to the single crystal structure determination, Li3InP2 crystallizes in the tetragonal space group $I4_1/acd$ (no. 142) with seven independent crystallographic positions (one for In, three each for Li and P; Table S2†). Considering that the crystal structure is based on a tetragonally distorted cubic close packing of phosphorus atoms, the multiplicity of the phosphorus Wyckoff positions (32g + 16e + 16i) leads to a total of 128 tetrahedral voids and 64 octahedral voids. One quarter of these tetrahedral voids is filled by the indium atoms (Wyckoff position 32g). The remaining 96 tetrahedral voids are occupied by lithium ($3 \times 32g$). Hence, the tetrahedral voids are fully occupied, whereas all octahedral voids are empty. The unit cell determined by single crystal X-ray diffraction is displayed in Fig. 1a.

Indium and phosphorus form InP4 tetrahedra, and four corner-sharing InP4 tetrahedra build a T2-supertetrahedron. These T2-supertetrahedra are interconnected via corners, yielding two independent adamantine-like networks, which are shown in red and blue colors in Fig. 1a and c.

The In and P atoms are covalently connected to four and two atoms, respectively, resulting in a formal negative charge for both In and P of (−1). Since the P atoms at the corner of the supertetrahedra are shared with the next supertetrahedron, one such unit can be written as $[\text{In}_3\text{P}_6\text{P}_4]^{12-}$ (Fig. 1b), which leads to an electronically balanced formula Li3InP2 ($\approx$Li12 [In3P6P4]12− or Li13In12P4).

The InP4 units slightly deviate from an ideal tetrahedron with P–In–P angles ranging from 107.20(1) to 111.55(1)°. The bond lengths within the InP4 tetrahedra are in the narrow range between 2.5676(5) and 2.5899(5) Å and are very similar to those in compounds with strong In–P interactions like InP (2.5412(1) Å) and Na3InP2 (2.592(3)–2.682(3) Å) in excellent agreement with DFT computations after full structural optimization (2.57–2.58 Å). The Li–P bonds in Li3InP2 range from 2.526(2) to 2.673(2) Å and are in good agreement with those in other binary or ternary phases containing Li and P.6,10,12,13,15 DFT optimization yields 2.51–2.67 Å, again practically superimposable with the experimental results.

Considering each center of gravity of the supertetrahedra, the arrangement of the independent networks of the T2-supertetrahedra corresponds in a hierarchical relationship to the arrangement of the Cu and Fe cations in the chalcopyrite structure, which is highlighted in Fig. 1d. The concept of supertetrahedra is already known in the literature, including supertetrahedral sulfides,21,22 which show structures with huge cavities, and also phosphidosilicates.8,23

**MAS-NMR spectroscopy**

Li$^1$ and $^{31}$P MAS-NMR measurements (Fig. 2) support the results of the crystal structure determination. The $^3$Li NMR spectrum shows only one signal with a chemical shift of 3.85 ppm. As expected, the NMR experiment cannot distinguish between the three crystallographically different lithium atoms, all of which are tetrahedrally coordinated by phosphorus in a very similar
chemical environment. The chemical shift of the Li atoms is in the same range as those for related compounds like Li$_9$AlP$_4$ (4.2 ppm), Li$_3$AlP$_2$ (4.0 and 3.0 ppm), Li$_3$GaP$_2$ (4.1 and 3.4 ppm), Li$_3$SiP$_2$ (2.1 ppm from $^7$Li MAS-NMR spectroscopy), and Li$_2$GeP$_2$ (3.6 and 2.4 ppm). Compared to the above-mentioned compounds with two signals in the $^6$Li NMR spectrum, the difference in local coordination, which is expressed by the P–Li–P angles, is the lowest for Li$_3$InP$_2$ (Li$_3$InP$_2$: 104.99(8)–113.25(8)$^\circ$, Li$_3$AlP$_2$: 100.0(3)–116.647(1)$^\circ$, Li$_3$GaP$_2$: 102.258(1)–115.2(3)$^\circ$, Li$_2$GeP$_2$: 84.68(1)–158.89(2)$^\circ$). The $^{31}$P NMR spectrum displays a very broad, asymmetric signal in the range of −260 to −360 ppm. This range is typical for chemical shifts of two-fold

![Diagram](image-url)
connected $\text{P}^{3-}$ atoms such as in $\text{Li}_3\text{AlP}_2$ (−300 and −308.7 ppm) or $\text{Li}_3\text{GaP}_2$ (−234.8 and −280.5 ppm).\(^4\) However, the signals of two-fold connected $\text{P}^{3-}$ atoms in the related phosphidotetrelates are much more downfield shifted ($\text{Li}_2\text{SiP}_2$: −219.1 and −241.5 ppm and $\text{Li}_2\text{GeP}_2$: −59.9, −164.8 and −178.4 ppm) due to the deshielding of the more electronegative tetrel elements compared to indium.\(^5\)

Interestingly, only one $^{31}\text{P}$ NMR signal is observed for $\text{Li}_3\text{InP}_2$, whereas two signals are obtained for all other related compounds. This correlates with the fact that the smallest distortion of the $\text{E}–\text{P}–\text{E}$ bond angles is observed for $\text{Li}_3\text{InP}_2$ [106.411(9)–111.41(1)\(^\circ\)] compared to $\text{Li}_3\text{AlP}_2$ [78.298(1)–111.709(1)\(^\circ\)]. \(^6\)

**Comparison of $\text{Li}_3\text{InP}_2$ with the lighter homologues $\text{Li}_3\text{AlP}_2$ and $\text{Li}_3\text{GaP}_2$**

Recently, we described the two isotypic phases $\text{Li}_3\text{AlP}_2$ and $\text{Li}_3\text{GaP}_2$,\(^7\) which crystallize in a distorted orthorhombic packing of phosphorus atoms in the space group $\text{Cmce}$ (no. 64) with lattice parameters $a = 11.5138(2)$, $b = 11.7634(2)$, $c = 5.8202(1)$ Å and $a = 11.5839(2)$, $b = 11.7809(2)$, $c = 5.8129(2)$ Å, respectively, both determined by Rietveld refinement at room temperature. The crystal structures are built up by corner- and edge-sharing $\text{TrP}_4$ ($\text{Tr}$ = Al, Ga) tetrahedra in two-dimensional layers. Based on a close packing of P atoms, the lithium atoms are located in all tetrahedral voids (Fig. 3). By contrast, $\text{Li}_3\text{InP}_2$ crystallizes in a tetragonal distorted phosphorus lattice in the space group $\text{I}4_1/\text{acd}$ (no. 142) with lattice parameters of $a = 12.03049(8)$ and $c = 23.9641(3)$ Å, determined by Rietveld refinement at room temperature, and as mentioned above, the single crystal structure determination reveals a three-dimensional structure with exclusively corner-sharing $\text{InP}_4$ tetrahedra for t-$\text{Li}_3\text{InP}_2$ (Fig. 1).

In order to gain additional insight into the experimentally observed structure types, we performed DFT-based structural optimizations for the Al, Ga and In compounds using the PBEsol functional\(^14\) as implemented in CASTEP\(^25\) (computational details are given in the ESI). In addition to the experimentally determined unit cells we performed a substitutional cross-check: both modifications, orthorhombic o-$\text{Li}_3\text{TrP}_2$ and tetragonal t-$\text{Li}_3\text{TrP}_2$, were used for $\text{Tr} = \text{Al}$, Ga and In, starting either from the experimentally determined structure or from a hypothetical one obtained by substituting the Tr species. The DFT-optimized cell parameters are in excellent agreement with the experiment for the title compound (we obtained $a_{\text{DFT}} = 11.96$ Å and $c_{\text{DFT}} = 23.74$ Å; full results are listed in Table S7,\(^\dagger\) Fig. 4a shows the resulting energies, relative to the respective binary phosphides similar in spirit to our recent work on $\text{Li}_3\text{AlP}_2$.\(^8\) We compute the DFT electronic energy, $E$, for the relaxed ternary structure as well as for $\text{Li}_3\text{P}$ and the respective zinc blende-type phase of $\text{AlP}$, $\text{GaP}$ or $\text{InP}$; the difference (in the sense of a “reaction energy”) then allows us to estimate the stability of the ternary phase:

$$\Delta E = E(\text{Li}_3\text{TrP}_2) - [E(\text{Li}_3\text{P}) + E(\text{TrP})]$$

Negative values of $\Delta E$ therefore indicate that the ternary phase is stable with respect to the binaries (Fig. 4a).

The compounds $\text{Li}_3\text{TrP}_2$ are energetically favored over their respective binary components $\text{Li}_3\text{P}$ and $\text{AlP}$, $\text{GaP}$ and $\text{InP}$. The latter all adapt the cubic zinc blende type. The energy gain is significant considering the known stability of the zinc blende type that is most frequent among III–V semiconductors. More importantly, the difference in pairs of $\Delta E$ values allows us to compare the tendency for assuming either the $\text{Cmce}$ or the $\text{I}4_1/\text{acd}$ structure for all of the $\text{Li}_3\text{TrP}_2$ phases. For the Al and Ga compounds, the $\text{Cmce}$ structure is favored by about 0.06 and 0.03 eV per formula unit (f.u.), respectively; by contrast, the $\text{I}4_1/\text{acd}$ structure is preferred for $\text{Li}_3\text{InP}_2$ (by about 0.06 eV f.u.\(^{-1}\)) in all agreement with experiments. The stabilization of the title compound compared to the constituent binary phosphides is computed to be 0.31 eV f.u.\(^{-1}\) (indicated by a negative sign in the convention of Fig. 4a), which represents a significant gain in stability and explains the synthetic accessibility of the ternary compound. Whilst there will always remain a certain error due

---

**Fig. 3** Structural details of $\text{Li}_3\text{AlP}_2$: (a) layers of $^3\text{[AlP}_{4}^{3-}]$ polyanions are separated by lithium ions. (b) Top view of one $^3\text{[AlP}_{4}^{3-}]$ layer of dimers of two edge-sharing $\text{AlP}_4$ tetrahedra that are interconnected by vertex-sharing. Unfilled voids within the layer are filled with additional Li ions. $\text{AlP}_4$ tetrahedra are shown as blue polyhedra. Al, P and Li atoms are drawn in orange, purple and grey, respectively.
Fig. 4 First-principles DFT computations for Li₃AlP₂, Li₃GaP₂ and Li₃InP₂: (a) Computed total energies for DFT-relaxed structural models of the Li₃TrP₂ (Tr = Al, Ga, or In) crystal structures, from a hypothetical, DFT-generated, Cmce Li₃InP₂ structure, emphasizing the edge-sharing [InP₄] tetrahedra, which may be compared to (c) the DFT-optimized $I4_1/acd$ Li₃InP₂ structure, from which a supertetrahedral fragment is shown.

Comparison of Li₃InP₂ with the phosphidotetrelates Li₂SiP₂ and Li₂GeP₂

The crystal structure of Li₃InP₂ is related to the structure of Li₂SiP₂ and Li₂GeP₂. The two latter isotypic phases also crystallize in the space group $I4_1/acd$ (no. 142), with lattice parameters of $a = 12.1111(1)$ and $c = 18.6299(4)$ Å for Li₂SiP₂ and $a = 12.3070(1)$ and $c = 19.0307(4)$ Å for Li₂GeP₂ and a slightly longer $a$, but much shorter $c$ parameter as compared to Li₃InP₂. A full comparison of the lattice parameters and the tetrahedral volumes in Li₂SiP₂, Li₂SiP₂ and Li₂GeP₂ is given in Table 2.

Assuming an average volume of 18 Å³ per heavy atom, the increase in cell volume corresponds approximately to the volume of 32 additional lithium atoms in the unit cell of Li₂SiP₂. Besides the change in the number of Li atoms, also the larger volume of the InP₄ tetrahedra compared to SiP₄/GeP₄ (see Table 2) contributes to an overall increase of the volume. However, the trends of the interatomic Tr-Tr (Tr = Al, Ga, In) distances in Li₃AlP₂, Li₃GaP₂ and Li₃InP₂ are listed in Table 1. Regarding the different orthorhombic (Li₃AlP₂, Li₃GaP₂) and tetragonal structures (Li₃InP₂), the interatomic distances of the metal atoms are shorter in the orthorhombic structures, where edge-sharing tetrahedra occur compared to the tetragonal structure, where only corner-sharing tetrahedra are present. One might ask for the origin of the preference of one structure type over the other when comparing all three phosphidotetrelates side-by-side. Interestingly, the results of the calculations are in agreement with Pauling’s third rule. At least qualitatively and within the limits of such empirical concepts, edge-sharing tetrahedra are disfavored on account of the repulsion of positively charged central atoms (Fig. 4b and c). This effect might be expected to be strongest in the In compound, where not only the ionic radius is the largest of the three, but the computed Mulliken charges for the series of Cmce structures (Al: +0.42e, Ga: +0.57e, hypothetical In structure: +0.65e) appear to be consistent with an increasing repulsion of Tr atoms in the case of edge-sharing tetrahedra. Note that the Mulliken charges, derived from quantum-mechanical computation, are not to the same as the formal negative charge of the Tr atom using the Lewis valence model (Fig. 1b). Accordingly, a structure containing edge-sharing tetrahedra is observed for Li₃AlP₂ and Li₃GaP₂, but not for Li₃InP₂ (Fig. 4c). This trend of the differences of the different metal to metal distances by DFT calculation is confirmed by the experimental interatomic Tr-Tr (Tr = Al, Ga, In) distances (Table 1). The experimental In-In distance is significantly longer than the Al-Al or Ga-Ga distances (4.116(3) Å vs. 3.028(5) Å (Al) and 3.089(2) Å (Ga)).

Table 1 Comparison of the shortest interatomic Tr-Tr (Tr = Al, Ga, In) distances, the cell volume and the volume per formula unit in Li₃AlP₂, Li₃GaP₂ and Li₃InP₂, from Rietveld refinements at room temperature

<table>
<thead>
<tr>
<th></th>
<th>Li₃AlP₂</th>
<th>Li₃GaP₂</th>
<th>Li₃InP₂</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cell volume/Å³</td>
<td>788.29(2)</td>
<td>793.28(2)</td>
<td>3468.39(6)</td>
</tr>
<tr>
<td>Volume per formula unit/Å³</td>
<td>98.54</td>
<td>99.16</td>
<td>108.38</td>
</tr>
<tr>
<td>Tetrahedron volume/Å³</td>
<td>7.0897</td>
<td>7.1334</td>
<td>8.8857</td>
</tr>
<tr>
<td>Tr-Tr distances/Å</td>
<td>3.028(5)</td>
<td>3.089(2)</td>
<td>4.116(3)</td>
</tr>
</tbody>
</table>
this increase is highly anisotropic, since in Li$_3$InP$_2$ the lattice parameter $c$ increases strongly, whereas the lattice parameter $a$ is even slightly shorter compared to the one in Li$_2$SiP$_2$ and Li$_2$GeP$_2$.

Fig. 5 shows a comparison of the structures of Li$_3$InP$_2$ and Li$_2$SiP$_2$ viewed along the $a$ and $c$ direction. In Li$_3$InP$_2$ the InP$_4$ tetrahedra respectively the T2-supertetrahedra are aligned in an almost parallel fashion, whereas in Li$_2$SiP$_2$ the T2-supertetrahedra are rotated along the tetragonal axes. Interestingly, the parallel alignment in Li$_3$InP$_2$ leads to a slight decrease of the $a$ and $b$ axes despite the higher lithium content, but to a significant increase of the $c$ axes.

In Table 3 the Wyckoff positions in Li$_3$InP$_2$ and Li$_2$SiP$_2$ are compared (Li$_2$GeP$_2$ is omitted since it is isotypic to Li$_2$SiP$_2$). The higher Li content of the In compound arises from the occupation of two 32g Wyckoff sites instead of two 16f sites in the tetrelates. As a consequence, the coordination environments of the lithium atoms in the structures are different. The coordination of the lithium atoms in Li$_3$InP$_2$ and Li$_2$SiP$_2$ by phosphorus is illustrated in Fig. S2 and S8, respectively. The positions Li1 and Li3 are similarly coordinated by four phosphorus atoms forming a distorted tetrahedron. By contrast, Li2 fills a strongly distorted octahedral void of phosphorus atoms with significant longer Li–P distances compared to Li1 and Li3.

---

**Table 2** Comparison of the tetrahedral volumes of InP$_4$, SiP$_4$ and GeP$_4$, the distances and angles between the supertetrahedra and the lithium coordination in Li$_3$InP$_2$, Li$_2$SiP$_2$ and Li$_2$GeP$_2$ obtained from single crystal structure determination. The values for Li$_2$SiP$_2$ and Li$_2$GeP$_2$ are taken from the literature.

<table>
<thead>
<tr>
<th>Atom</th>
<th>Li$_3$InP$_2$</th>
<th>Li$_2$SiP$_2$</th>
<th>Li$_2$GeP$_2$</th>
</tr>
</thead>
<tbody>
<tr>
<td>a/Å</td>
<td>12.0007(2)</td>
<td>12.111(1)</td>
<td>12.3070(1)</td>
</tr>
<tr>
<td>c/Å</td>
<td>23.917(5)</td>
<td>18.6299(4)</td>
<td>19.0307(4)</td>
</tr>
<tr>
<td>V/Å$^3$</td>
<td>3447.71(1)</td>
<td>2732.61(7)</td>
<td>2882.42(9)</td>
</tr>
<tr>
<td>$V(T2/1)P4$ tetrahedra/Å$^3$</td>
<td>8.7944</td>
<td>5.8042</td>
<td>6.4083</td>
</tr>
</tbody>
</table>

Distances (Å) between the centers of the T2-supertetrahedra in Li$_3$InP$_2$, Li$_2$SiP$_2$ and Li$_2$GeP$_2$.

---

**Table 3** Comparison of Wyckoff positions and atomic coordinates in Li$_3$InP$_2$ and Li$_2$SiP$_2$.

<table>
<thead>
<tr>
<th>Atom</th>
<th>Wyck.</th>
<th>x</th>
<th>y</th>
<th>z</th>
</tr>
</thead>
<tbody>
<tr>
<td>In1/Si</td>
<td>32g</td>
<td>0.11730(2)</td>
<td>0.12523(2)</td>
<td>0.31365(2)</td>
</tr>
<tr>
<td>P1</td>
<td>16d</td>
<td>0</td>
<td>1/4</td>
<td>0.00063(2)</td>
</tr>
<tr>
<td>P2</td>
<td>16e</td>
<td>0.23848(3)</td>
<td>0</td>
<td>1/4</td>
</tr>
<tr>
<td>P3</td>
<td>32g</td>
<td>0.25031(2)</td>
<td>0.25604(2)</td>
<td>0.12516(2)</td>
</tr>
<tr>
<td>Li1</td>
<td>32g</td>
<td>0.1279(2)</td>
<td>0.3720(2)</td>
<td>0.0615(1)</td>
</tr>
<tr>
<td>Li2</td>
<td>32g</td>
<td>0.1379(2)</td>
<td>0.1250(2)</td>
<td>0.0651(1)</td>
</tr>
<tr>
<td>Li3</td>
<td>32g</td>
<td>0.3635(2)</td>
<td>0.1212(2)</td>
<td>0.19053(9)</td>
</tr>
</tbody>
</table>

In Table 3 the Wyckoff positions in Li$_3$InP$_2$ and Li$_2$SiP$_2$ are compared (Li$_2$GeP$_2$ is omitted since it is isotypic to Li$_3$InP$_2$). The higher Li content of the In compound arises from the occupation of two 32g Wyckoff sites instead of two 16f sites in the tetrelates. As a consequence, the coordination environments of the lithium atoms in the structures are different. The coordination of the lithium atoms in Li$_3$InP$_2$ and Li$_2$SiP$_2$ by phosphorus is illustrated in Fig. S2 and S8, respectively. The positions Li1 and Li3 are similarly coordinated by four phosphorus atoms forming a distorted tetrahedron. By contrast, Li2 fills a strongly distorted octahedral void of phosphorus atoms with significant longer Li–P distances compared to Li1 and Li3.
Here, the lithium atom Li2 is not located in the center of gravity of the octahedron but shows two much longer distances to neighboring P atoms of the distorted octahedron, resulting in a butterfly-type coordination of four P atoms. Interestingly, despite the smaller amount of Li atoms in Li2SiP2, not all the tetrahedral voids are occupied. In both compounds 25% of the tetrahedral voids are occupied by In or Si. Whereas all of the remaining 75% tetrahedral voids in Li3InP2 are filled with Li, only 37.5% are occupied by Li in Li2SiP2. In the latter, however, Li atoms occupy 25% of the distorted octahedral voids.

The different occupation of voids in Li3InP2 and Li2SiP2 also results in a different coordination of the supertetrahedra by lithium, which is shown in Fig. 6.

The different Li coordination arises from the different charges of the supertetrahedra Si4P8−/Ge4P8− and In4P812− (Fig. 1b). In Li3InP2 the lithium atoms form an almost regular octahedron around the indium atom with In–Li distances in the narrow range of 3.041 to 3.131 Å with an average of 3.075 Å, whereas in Li2SiP2, the octahedron formed by lithium atoms around silicon is strongly distorted with longer average distances of 3.222 Å and values between 2.958 and 3.556 Å. As a consequence, also octahedral voids of P atoms are filled with Li ions in Li2SiP2.

**Impedance spectroscopy**

For Li3InP2 two impedance measurements were performed to determine the ionic conductivity. The results are shown in Fig. S10.† The semi-circle can be described as parallel circuit element of a resistor and a constant phase element (R/Q). For the constant phase element the fit of the data acquired at 298 K resulted in a values of $\approx 0.99$ and $Q$ parameters of $\approx 2 \times 10^{-8}$ F s$^{(a-1)}$. The conductivity was determined to $\sigma_{\text{Li3InP2}} = 2.8(2) \times 10^{-9}$ S cm$^{-1}$ at 298 K (obtained from two independently measured cells). DC polarization measurements in the range from 50 to 150 mV reveal an electronic conductivity of $2.7(3) \times 10^{-9}$ S cm$^{-1}$ at 298 K (based on the standard deviation of two cells). The conductivity value obtained by DC polarization measurements is in the same range as the value obtained by PEIS measurements. Hence, the Nyquist plot shows only the semi-circle of the electronic conductivity, and no semi-circle for the ionic conductivity appears.

**Conclusions**

Li3InP2 is the first lithium phosphidoindate and can be described as a tetragonally distorted fcc lattice of P atoms (space group $I4_1/acd$), in which the In atoms occupy tetrahedral voids, thus forming a polyananionic framework of In4P8 supertetrahedra. The lithium atoms occupy the remaining tetrahedral voids. The structure of the compound is not isotypic to the previously reported ones of the lighter homologues, the orthorhombic compounds Li3AlP2 and Li3GaP2 (space group $Cmce$), which feature 2D layers of connected tetrahedra. First-principles DFT computations confirm the trend for the Al and Ga (In) compounds to crystallize in the orthorhombic (tetragonal) structure, respectively, which might originate in the different repulsive cation···cation interactions in both structures. Impedance spectroscopy reveals a very low electronic, but no...
ionic conductivity, whereas Li$_2$SiP$_2$ and Li$_2$GeP$_2$ show a moderate ionic mobility ($2.2(3) \times 10^{-7}$ S cm$^{-1}$ at 293 K and 1.5(3) $\times 10^{-7}$ S cm$^{-1}$ at 300 K, respectively). The geometrical analysis of the Li positions shows that in Li$_3$InP$_2$ all tetrahedral voids are fully occupied by lithium, whereas in Li$_3$SiP$_2$ and Li$_3$GeP$_2$ tetrahedral voids remain empty, and especially strongly distorted octahedral sites are filled. In accordance with the observations in few phosphide-based lithium ion conductors such as Li$_4$AlP$_2$, lithium diffusion preferably appears on pathways via partially occupied octahedral sites.

Overall, these results demonstrate that even though crystal structures of phosphide compounds can contain complex poly-anionic networks, a relatively simple description in terms of distorted close-packed arrangements of phosphorus atoms gives better insight for the description of lithium ion mobility. The title compound Li$_3$InP$_2$ provides a missing link in two respects: (i) it shows the structure changes in the series Li$_3$TrP$_2$ for $\text{Tr} = \text{Al, Ga, In}$, and (ii) it shows changes in lithium mobility in the series Li$_3$InP$_2$, Li$_3$SiP$_2$ and Li$_3$GeP$_2$.

**Author contributions**

TMFR carried out the crystal structure determination by single crystal and powder X-ray diffraction, performed the impedance spectroscopy measurements and wrote the manuscript draft. VLD carried out the DFT computations and provided discussion. JM contributed to the synthesis and data evaluation. GRS performed NMR experiments. TF designed research, provided guidance, and critically reviewed the manuscript.

**Conflicts of interest**

The authors declare no competing financial interest.

**Acknowledgements**

The work was carried out as part of the research project ASSB coordinated by ZAE Bayern. The project is funded by the Bavarian Ministry of Economic Affairs, Regional Development and Energy. V. L. D. acknowledges a Leverhulme Early Career Fellowship. The authors would like to acknowledge the use of the University of Oxford Advanced Research Computing (ARC) facility in carrying out this work (see DOI: 10.5281/zenodo.22558).

**References**