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Structure modulation of g-C₃N₄ in TiO₂{001}/g-C₃N₄ hetero-structures for boosting photocatalytic hydrogen evolution†

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Structure design of photocatalysts is highly desirable for taking full advantage of their abilities for H₂ evolution. Herein, the highly-efficient TiO₂{001}/g-C₃N₄ (TCN) heterostructures have been fabricated successfully *via* an *in situ* ethanol-thermal method. And the structure of g-C₃N₄ in the TCN heterostructures could be exfoliated from bulk g-C₃N₄ to nanosheets, nanocrystals and quantum dots with the increase of the synthetic temperature. Through detailed characterization, the structural evolution of g-C₃N₄ could be attributed to the enhanced temperature of the ethanol-thermal treatment with the shear effects of HF acid. As expected, the optimal TCN-2 heterostructure shows excellent photocatalytic H₂ evolution efficiency (1.78 mmol h⁻¹ g⁻¹) under visible-light irradiation. Except for the formed built-in electric field, the significantly enhanced photocatalytic activity of TCN-2 could be ascribed to the enhanced crystallinity of TiO₂{001} nanosheets and the formed g-C₃N₄ nanocrystals with large surface area, which could extend the visible light absorption, and expedite the transfer of photo-generated charge carriers further. Our work could provide guidance on designing TCN heterostructures with the desired structure for highly-efficient photocatalytic water splitting.

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1. Introduction

Semiconductor photocatalysis provides an efficient strategy for overcoming the energy crisis and global climate deterioration by utilizing solar energy to generate sustainable hydrogen (H₂) energy from water splitting.^{1–3} Since Fujishima discovered TiO₂ could split water into H₂ in 1972, TiO₂ has been paid much attention because of its desirable properties such as low cost, non-biotoxic, photo-corrosion resistance and high photocatalytic activity.^{4,5} To date, extensive research has verified that the tailored nanocrystal facets of TiO₂ have a close relationship with its photocatalytic activity of H₂ evolution. Due to the high surface energy (0.90 J m⁻²), the {001} facet of anatase TiO₂ shows higher reactive activity than the commonly investigated {101} facet.^{6,7} However, the practical application of TiO₂ {001} is still impeded by the ineffective utilization of solar light caused by the wide bandgap (~3.2 eV) and the high recombination rate of photo-generated e⁻-h⁺ pairs.^{8–10} Therefore, exploring the high efficient visible-light driven TiO₂{001} based photocatalyst is still highly desired for water splitting.

Admittedly, incorporating TiO₂{001} with other narrow bandgap semiconductors is supposed to overcome the intrinsic drawbacks of pristine TiO₂{001} by forming the internal electric

field.^{11,12} Owing to the excellent chemical stability, tunable electronic structure, low-cost and narrow bandgap (~2.7 eV), the metal-free g-C₃N₄ has been identified as a promising candidate for fabricating the heterostructures with TiO₂{001}.^{13,14} Benefiting from the broadened spectral absorption range and the facilitated spatial separation of photo-generated charge carriers, constructing the high-efficient TiO₂{001}/g-C₃N₄ (TCN) heterostructure is always a hot issue for the photocatalytic H₂ evolution.¹⁵ For instance, Tan *et al.* vapor-deposited g-C₃N₄ nanoparticles on the commercialized P25, leading to the rapid migration rate of photo-generated charge carriers.¹⁶ However, the commonly employed bulk g-C₃N₄ (BCN) also suffers from the instinct drawbacks of small specific surface area and high recombination rate of photo-generated e⁻-h⁺ pairs, resulting in the limited practical application of TiO₂/BCN heterostructures.^{30–32} It has been verified that the structure of g-C₃N₄ has a substantial impact on the electronic energy level and surface properties of the fabricated TCN heterostructures.^{20–22} Coupling TiO₂ with a target-specific structure of g-C₃N₄ has been employed to further improve the photocatalytic H₂ production performance. Wang *et al.* coupled the g-C₃N₄ nanosheets (CN-NS) with TiO₂ flower, showing high efficient visible-light photocatalytic activity.¹⁷ Guo *et al.* hybridized the TiO₂ nanoparticles with g-C₃N₄ nanocrystalline (CN-NC), showing the extended lifetime of photo-generated e⁻-h⁺ pairs.¹⁸ Li *et al.* grafted the g-C₃N₄ quantum dots (CN-QDs) onto TiO₂ nanotube, causing the enhanced photocatalytic H₂ evolution performance.¹⁹ To date, numerous efforts have been

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implemented to fabricate the TCN heterostructures with a targeted structure of $g\text{-C}_3\text{N}_4$.^{23–26} However, it remains a severe challenge to precisely synthesize the TCN heterostructures with different structures of $g\text{-C}_3\text{N}_4$ via easily modulating one synthetic parameter. And few studies concerning the influence of the morphological structure of $g\text{-C}_3\text{N}_4$ its-self on the photocatalytic H_2 evolution performance of the fabricated TCN heterostructures. In addition, the formation of intimate interfacial contact between $\text{TiO}_2\{001\}$ and $g\text{-C}_3\text{N}_4$ is crucial for preparing the TCN heterostructures with superior photocatalytic performances.^{27–29} Compared with the widely used mechanical blend and unordered sintering methods, the *in situ* solvothermal method could effectively form the strong chemical bond between TiO_2 and $g\text{-C}_3\text{N}_4$, which promotes the formation of the close contact interface.^{33–35}

Directed by the above strategy, we aimed to synthesize the TCN heterostructure via the *in situ* ethanol-thermal method for the high-efficient photocatalytic H_2 evolution. And the structure of $g\text{-C}_3\text{N}_4$ in the prepared TCN heterostructures could be modulated from BCN to CN-NS, CN-NC and CN-QD via regulating the synthetic temperature. Through detailed characterizations, the structure evolution of $g\text{-C}_3\text{N}_4$ in TCN heterostructures could be attributed to the increased temperature of ethanol-thermal treatment and the shear effects of HF acid. The optimized $\text{TiO}_2/g\text{-C}_3\text{N}_4$ nanocrystalline (TCN-2) was proved to have the highest H_2 evolution performance under visible light irradiation. Except for the formed built-in electric field, the significantly enhanced photocatalytic activity of TCN-2 could be ascribed to the enhanced crystallinity of $\text{TiO}_2\{001\}$ and the formation of CN-NC with large surface area, which could extend the visible light absorption, and expedite the transfer of photo-generated charge carriers further. Our work might provide a novel path for the morphological structure designing of TCN heterostructures towards high-efficient photocatalytic H_2 evolution.

2. Experiment

2.1. Materials

Tetrabutyl titanate (TBOT), hydrofluoric acid (HF, 30 vol%), *tert*-butanol (TBA), triethanolamine (TEOA), ethanol (EtOH) and methanol (MeOH) were purchased from Aladdin-Holdings Group. Melamine, silver nitrate (AgNO_3), EDTA disodium salt dihydrate (EDTA-2Na), sodium sulfate (Na_2SO_4), chloroplatinic acid (H_2PtCl_6) and benzoquinone (PBQ) were supplied by Shanghai Macklin Biochemical Co., Ltd.

2.2. Synthesis of TCN-1, TCN-2 and TCN-3

BCN was synthesized by the traditional thermal polycondensation method by using melamine as the precursor.³⁶ Briefly, 6 g melamine was put into a closed crucible and calcinated at 550°C for 4 h with a ramp rate of 5°C min^{-1} directly. The light-yellow BCN powder was obtained after the fully ground.

TCN heterostructures were fabricated via an *in situ* ethanol-thermal approach. Briefly, 0.5 g of BCN was well dispersed into 30 mL of ethanol under magnetic stirring for 30 min. Then,

5 mL of $\text{Ti}(\text{O}i\text{Bu})_4$ and 1.5 mL of HF acid were successively added into the above suspension. After stirring for 15 min, the above mixture was transferred into a Teflon inner container with a capacity of 50 mL. Afterwards, the fully sealed high-pressure hydrothermal reactor was heated at the set temperature for 24 h. After cold to the ambient temperature, the sediments were washed thoroughly using ethanol and deionized (DI) water several times alternately until the pH became neutral. The final samples were obtained after vacuum drying at 60°C overnight. The morphological structure of $g\text{-C}_3\text{N}_4$ in the fabricated TCN heterostructures could be tuned from BCN to CN-NS, CN-NC and CN-QD via regulating the set temperature at 160°C , 200°C and 240°C , respectively.

2.3. Characterization

The morphological structures and chemical element compositions of the samples were observed by a high-resolution transmission electron microscope (HR-TEM, JEM-2100, Japan) and a field-emission scanning electron microscope (FE-SEM, Thermo Scientific GX4, USA) equipped with the energy dispersive spectroscopy (EDS) system. The crystalline structures were characterized by an X-ray diffractometer (XRD, Rigaku SmartLab, Japan) with $\text{Cu-K}\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$) with a range of 5° to 80° . The chemical bonding states were detected on a Fourier transform infrared spectrometer (FT-IR, Thermo Scientific NICOLET 6700, USA) between 3900 and 400 cm^{-1} using the standard KBr pellet technique. The surface oxidation states were recorded by an X-ray photoelectron spectroscopy instrument (XPS, Thermo Fisher ESCALAB Xi+, UK) with $\text{Al-K}\alpha$ radiation. The BET-specific surface areas was obtained on an N_2 micropore chemisorption analyzer (BET, Quantachrome, USA). The optical absorption properties were measured by a UV-vis spectrophotometer (UV-vis DRS, Agilent Carry 5000, AUS), using BaSO_4 as a reference. The photoluminescence (PL) emission spectra and transient fluorescence spectra were tested on a fluorescence spectrometer (Edinburgh FLS1000, UK) with a incident wavelength of 387 nm.

2.4. Electrochemical measurement

The photo-electrochemical (PEC) measurements were carried out on a CHI660E electrochemical workstation (Shanghai Chenhua, China) in $0.5 \text{ M Na}_2\text{SO}_4$ aqueous solution with a standard three-electrode system. The working electrode ($1.0 \times 1.0 \text{ cm}^2$) was prepared via the doctor-blading method.³⁷ And the saturated calomel electrode (SCE) and platinum (Pt) wire were served as the reference electrode and counter electrode, respectively. A 300 W xenon lamp equipped with a UV-cutoff filter ($\lambda > 420 \text{ nm}$) was used as the visible light source (CEL-HX300, Zhongjiao Jinyuan Technology Co., Ltd. China). Prior to measurement, the suspension was purged with N_2 to remove the residual air. The transient photocurrent response tests were performed at an applied -0.6 V potential versus SCE under continuously illumination ($\lambda > 420 \text{ nm}$) with the switch on and off every 40 s. The electrochemical impedance spectroscopy (EIS) Nyquist plots were recorded at open circuit voltage versus SCE over the frequency range from 1 MHz to 0.1 Hz in the dark.



2.5. Photocatalytic H₂ evolution tests

The photo-reduction activities of the prepared samples were evaluated by an online photocatalytic water splitting system (CEL-PAEM-D8, Zhongjiao Jinyuan Technology Co., Ltd. China). Briefly, 50 mg of sample was thoroughly dispersed into 80 mL of TEOA aqueous solution (10 vol%) under magnetic stirring for 30 min. As the precursor, 150 μ L of H₂PtCl₆ (3 wt%) aqueous solution was added to the above mixture for the subsequent *in situ* photo-deposition of Pt atoms on the prepared samples. The photocatalytic water splitting reaction was triggered under the above 300 W xenon lamp equipped with a UV-cutoff filter ($\lambda > 420$ nm). Before reaction, the whole system was vacuumed by a mechanical pump to remove any gas impurities. During the whole reaction process, the aqueous solution in the quartz reactor was maintained at 5 °C using a cooling water circulation system. The H₂ production rate was analyzed by a gas chromatograph (GC-2014C, Shimadzu).

2.6. Theoretical computation

The first-principles calculations were performed using the Vienna *Ab initio* Simulation Package (VASP) with the projector augmented wave (PAW) pseudo potentials. The generalized gradient approximation of Perdew–Burke–Ernzerhof parameterization (GGA-PBE) was applied to describe the exchange and correlation potential. The electronic wave functions use a cutoff energy of 400 eV, and $6 \times 4 \times 1$ Monkhorst–Pack *k*-point sampling was set for all calculations. Geometry optimizations were terminated when the energy and force on each ion were reduced below 10^{-8} eV and 0.01 eV \AA^{-1} , and the optimized structures were then used to calculate the electronic structures of the fabricated TCN heterostructures.

3. Results and discussion

As illustrated in Fig. 1, the TCN heterostructures with different structures of g-C₃N₄ have been synthesized *via* the *in situ*

ethanol-thermal method successfully. Firstly, BCN was synthesized by the thermal polymerization of melamine.³⁶ Then, the synthesized BCN powder was ultrasonically dispersed into the ethanol. Due to the theory of minimized mixing enthalpy, ethanol molecules would conquer the interlayer van der Waals force, and embed into the interlayer gallery of BCN.³⁸ Afterwards, the successively added TBOT molecules would anchor to the surface of CN-NS owing to the formation of coordinated bond between the Ti atom in TBOT and the N atom in g-C₃N₄, which provides a channel for the preferential transfer of photo-excited electrons.³⁹ Under the synergistic effect of the ethanol-thermal treatment and HF acid, the TBOT precursors would be *in situ* hydrolyzed to the anatase TiO₂{001} on the surface of g-C₃N₄. The g-C₃N₄ in the fabricated TCN heterostructures could be exfoliated from BCN to CN-NS (160 °C), CN-NC (200 °C) and CN-QD (240 °C) with the increasing of the synthetic temperature, respectively. In addition, the *in situ* ethanol-thermal method could provide the high temperature and high pressure condition, which is essential for the formation of high-crystalline TCN heterostructures with intimate contact.

3.1. The morphological structure evolution of g-C₃N₄

HR-TEM was used to observe the morphological structure of g-C₃N₄ in the fabricated TCN heterostructures. As shown in Fig. 2(a), the g-C₃N₄ in TCN-1 exhibits two-dimensional curved lamellar structure, which is in stark contrast to the thick morphology of BCN (shown in Fig. 2(e)), suggesting the successful exfoliation of BCN into CN-NS after the ethanol-thermal treatment at 160 °C for 24 h. And there are some TiO₂ nanosheets with an average diameter of 18 nm are homogeneously distributed on the surface of CN-NS (shown in Fig. S1†). Observed by HR-TEM (shown in Fig. 2(c)) further, the distinct lattice fringes with an interplanar spacing of 0.35 nm and 0.19 nm could be assigned to the (101) and the (002) facet of anatase TiO₂, respectively.^{40–42} It is well known that the (002) facet belongs to the {001} crystal plane, therefore the anatase TiO₂ with a high exposure of {001} facets has been synthesized

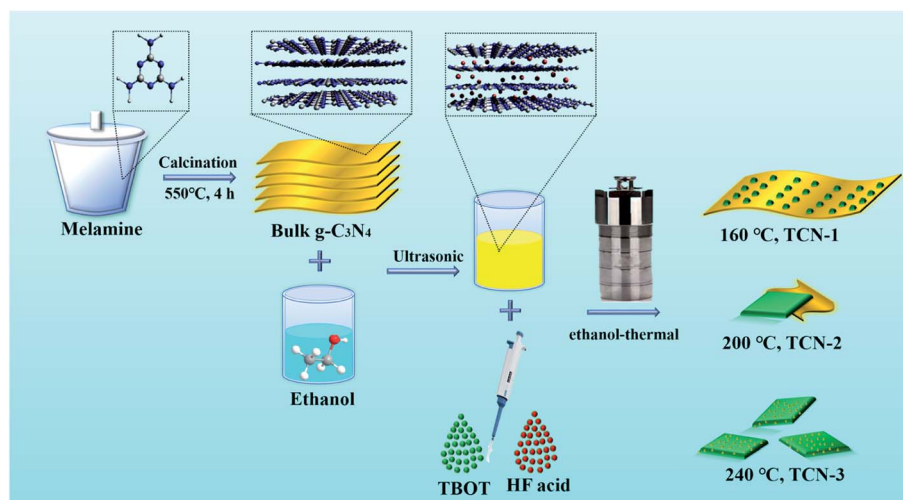


Fig. 1 Schematic illustration of TCN-1, TCN-2 and TCN-3 heterostructures fabricated at 160 °C, 200 °C and 240 °C, respectively.



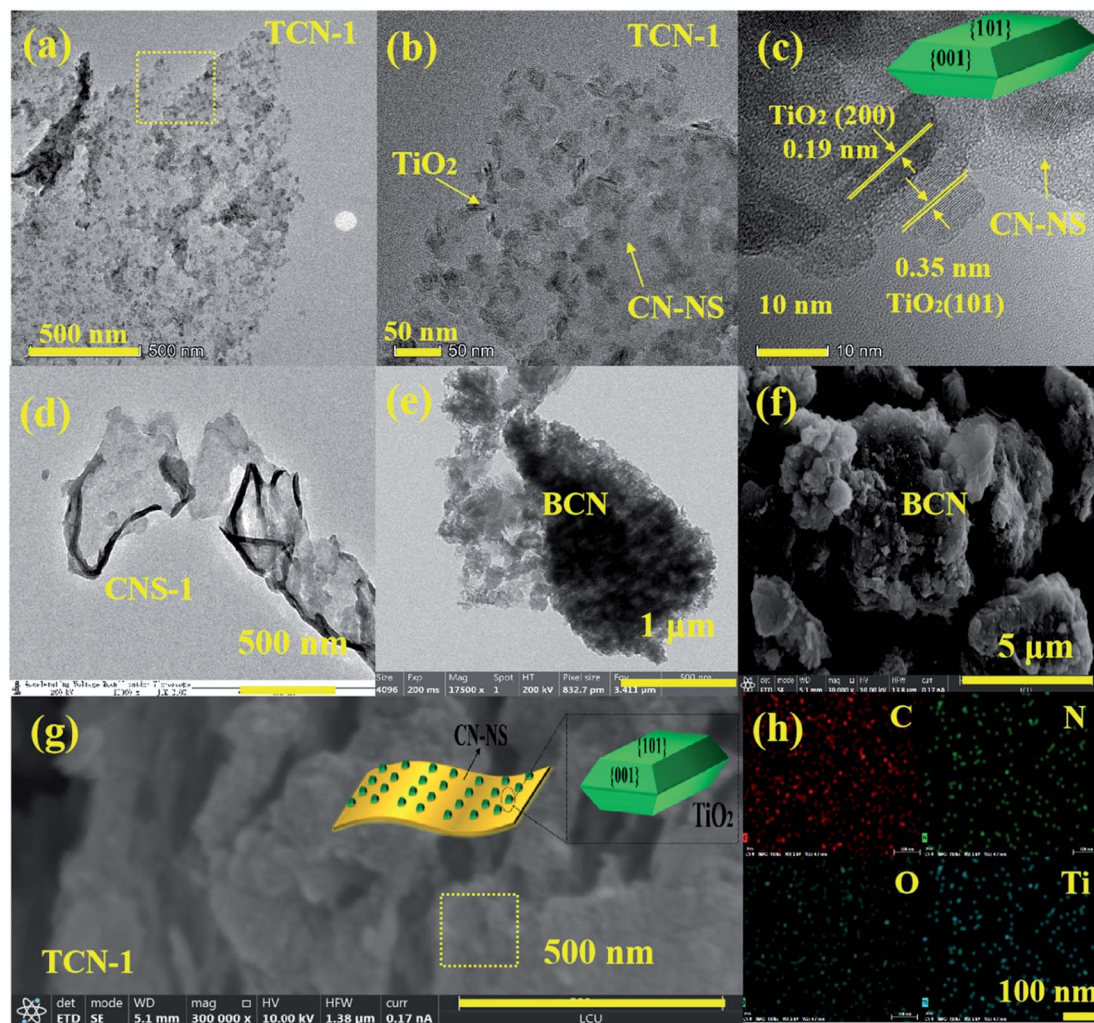


Fig. 2 (a) TEM, (b and c) HR-TEM, (g) FE-SEM and (h) EDS of TCN-1; (e) TEM and (f) FE-SEM of BCN; (d) TEM of pure CNS-1; the illustrated diagram of (the inset of (c)) $\text{TiO}_2\{001\}$ and (the inset of (g)) TCN-1.

successfully.^{43–45} And the formation mechanism of $\text{TiO}_2\{001\}$ facets has been discussed in our previous work.³⁶ Moreover, FE-SEM was employed to detect the morphology of the fabricated TCN-1 further. Consistent with the results of TEM, the lamellar structure of TCN-1 (shown in Fig. 2(g)) is different with the blocky structure of BCN (shown in Fig. 2(f)). In addition, the uniformly distributed C, N, Ti and O elements in the EDX mapping analysis of SEM (shown in Fig. 2(h)) also demonstrates the combined structure of g- C_3N_4 and TiO_2 in the fabricated TCN-1 heterostructures. Therefore, it can be reasonably deduced that the fabricated TCN-1 is composed of $\text{TiO}_2\{001\}$ and CN-NS, and its corresponding schematic diagram is shown in the inset of Fig. 2(g). To detect the exfoliation mechanism of BCN at 160 °C, BCN was treated at the same condition with TCN-1 except without adding the TBOT molecules. It is obvious in Fig. 2(d) that the formed CNS-1 also shows the typical two-dimensional curved lamellar structure. Thus, the successful exfoliation of BCN into CN-NS could be attributed to the ethanol-thermal treatment and the added HF acid.

It can be clearly observed in Fig. 3(a) that the large-area CN-NS vanished as the synthetic temperature increased to 200 °C. As shown in Fig. 3(b), the fabricated TCN-2 heterostructures have two distinctly different contrasts with similar sizes. Identified by the HR-TEM (shown in Fig. 3(c)), the light contrast with no distinct lattice fringes could be assigned to the ultrathin g- C_3N_4 nanosheet with small size, implying the successful exfoliation of BCN into CN-NC after the ethanol-thermal treatment at 200 °C for 24 h. And the dark fringes with the lattice spacing of 0.35 nm and 0.19 nm could be ascribed to the (101) and (002) planes of anatase TiO_2 , respectively, suggesting the successful synthesis of anatase TiO_2 with the exposure of {001} facets. Consistent with the results of TEM images, the SEM image of TCN-2 shown in Fig. 3(e) also shows a smaller lamellar structure in comparison with that of TCN-1 (shown in Fig. 2(f)). Combined with the results of EDS mapping shown in Fig. 3(f), it can be deduced that TCN-2 is constituted by TiO_2 and CN-NC, and its corresponding illustrated diagram is shown in Fig. 3(e). In addition, measured by the Nano measure, the average size of $\text{TiO}_2\{001\}$ is around 20 nm (shown in Fig. S6†),



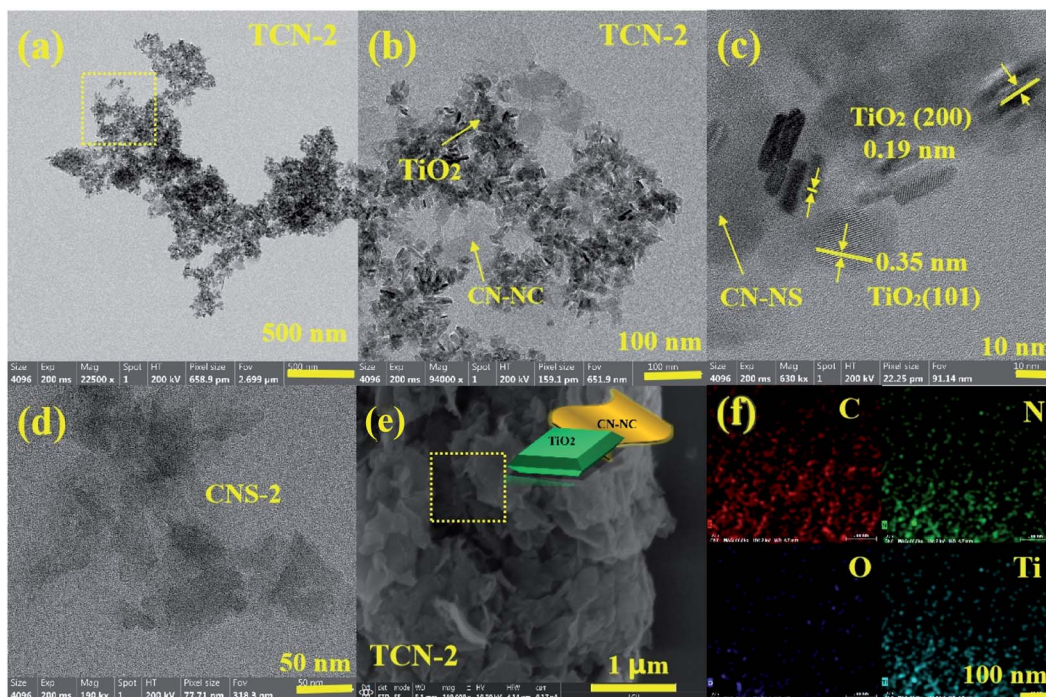


Fig. 3 (a and b) TEM, (c) HR-TEM, (e) SEM and (f) EDS mapping of TCN-2, (d) TEM of CNS-2 (the inset of (e)) the illustrated diagram of TCN-2.

and the lattice fringes of {001} facets in TCN-2 are more distinct than those in TCN-1, illustrating the improved crystallinity of TiO_2 {001} in TCN-2 due to the improved synthetic temperature. It has been proved that the enhanced crystallinity could

effectively improve the light absorption ability and reduce the number of recombination centers for the photo-generated e^- - h^+ pairs.⁴³ Therefore, the fabricated TCN-2 is expected to have a higher photocatalytic activity than TCN-1.

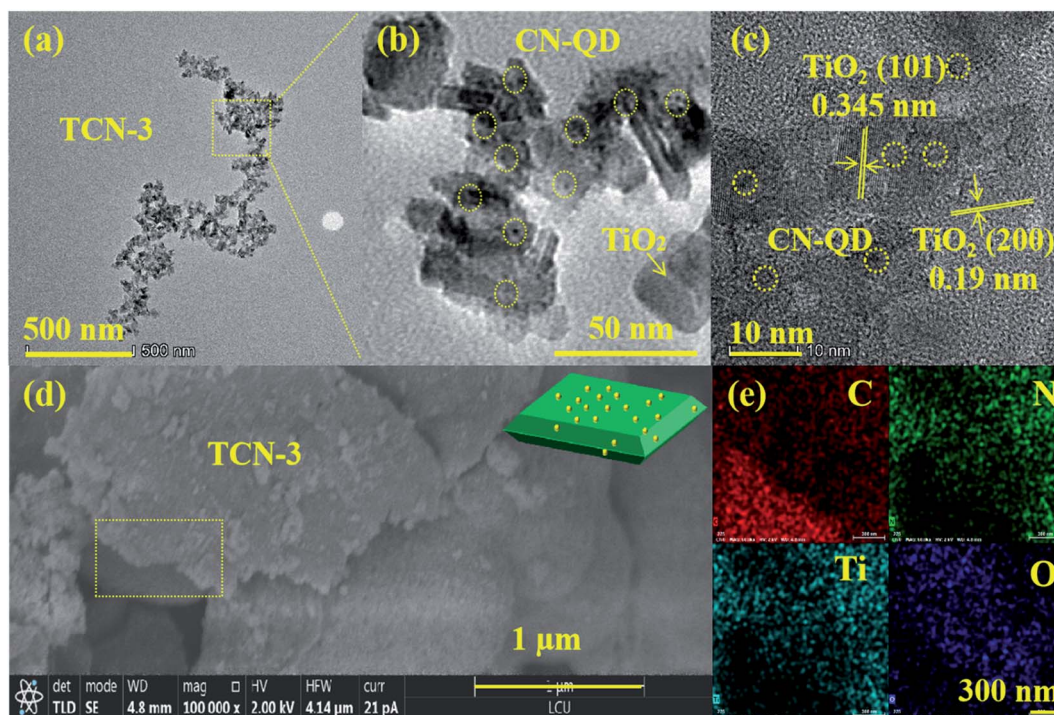


Fig. 4 (a and b) TEM, (c) HR-TEM and (d) SEM images of TCN-3 (the inset of (d)); the illustrated diagram of TCN-3; (e) EDX elemental mapping results of C, N, Ti and O in TCN-3.



To investigate the exfoliation mechanism of BCN at 200 °C, BCN was treated at the same condition with TCN-2 except without adding TBOT molecules. As shown in Fig. 3(d), the synthesized pure CNS-2 also shows the small sheeting structure. Therefore, it could be speculated that the successful exfoliation of BCN into CN-NC is attributed to the enhanced temperature of ethanol-thermal treatment and the shear effects of HF acid. Furthermore, to detect the effects of HF acid further, BCN was treated under the same condition with CNS-2 except without adding HF acid, the obtained sample was named as CN-200. It is apparent in Fig. S2† that the obtained CN-200 shows a typical layered structure with a large area. Therefore, the added HF acid is vital for breaking the C–N bonds between the adjacent tri-s-triazine rings under the enhanced synthetic temperature.

Once the synthetic temperature reached up to 240 °C, the ultra-thin CN-NC with small size is a disappeared (shown in Fig. 4(a)) and there are some uniform spherical quantum dots mono-dispersed on the surface of TiO₂ nanosheets (shown in Fig. 4(b)). Analyzed by Nano measure 1.2, the particle sizes of the formed quantum dots are range from 2 nm to 5 nm, and the average diameter is estimated to be 3.0 nm (shown in Fig. S3†). Although it is too small to observe the lattice fringe of the formed quantum dots, those spherical quantum dots could be assigned to the CN-QD according to the stability analysis of TiO₂ and g-C₃N₄.^{44,57} Moreover, the undetected morphological structure of BCN also confirms the successful exfoliation of BCN into CN-QD further. Investigated by HR-TEM (shown in Fig. 4(c)), the interplanar spacing of 0.19 nm could be assigned to the (200) planes of anatase TiO₂, suggesting the successful synthesis of anatase TiO₂ with the high exposure of {001} plane.

Measured by the Nano measure, the average size of TiO₂{001} is 23 nm (shown in Fig. S7†). Consistent with the results of TEM images, the lamellar structure of g-C₃N₄ is also disappeared in the SEM image of TCN-3 (shown in Fig. 4(d)), proving that the g-C₃N₄ in the synthesized TCN-3 has been exfoliated to the g-C₃N₄ quantum dots (CN-QD) further. For confirming the successful exfoliation of BCN into CN-QD at 240 °C further, the pure CN-QDs has been synthesized at the same condition with TCN-3 except without adding the TBOT molecules (shown in Fig. S4†). In addition, the EDX mapping analysis of SEM (shown in Fig. 4(e)) demonstrates the uniform distribution of C, N, Ti and O elements, confirming the combined structure of CN-QD and TiO₂ in TCN-3 heterostructures further. Therefore, it is reasonable to deduce that the exfoliated CN-QD have been successfully deposited on the surface of TiO₂{001}, and the corresponding illustrated diagram of TCN-3 is shown in the inset of Fig. 4(d).

3.2. Phase structures and composition

The crystal phase structures of the prepared samples were investigated by XRD patterns. As exhibited in Fig. 5(a), all the diffraction peaks of pristine TiO₂ could be well indexed to anatase TiO₂ (JCPDS No. 21-1272). The two pronounced diffraction peaks of BCN at 13.1° and 27.8° could be assigned to the (100) diffraction peak for the in-plane periodicity of continuous tri-s-triazine unites and the (002) diffraction plane of layered stacked π -conjugated aromatic rings, respectively (PDF #87-1526).⁴⁸ For TCN-1, all the characteristic diffraction planes of both anatase TiO₂ and g-C₃N₄ could be clearly

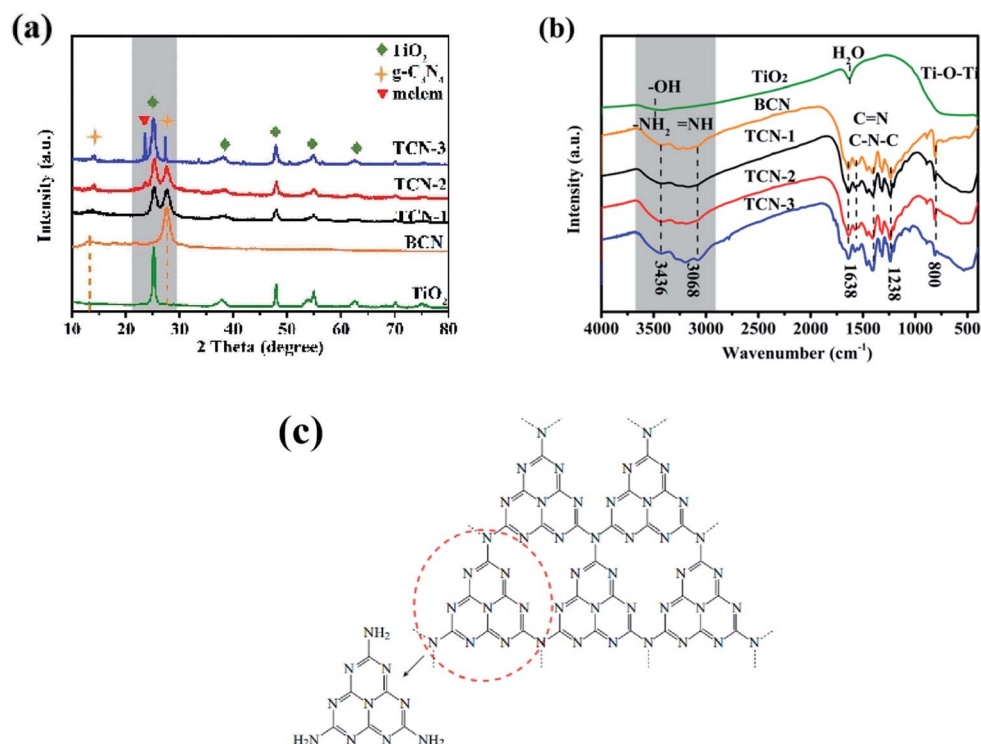


Fig. 5 (a) XRD patterns and (b) FT-IR spectra of TiO₂, BCN, TCN-1, TCN-2 and TCN-3; (c) the formation process of melem.



observed, corroborating the coexistence of anatase TiO_2 and $\text{g-C}_3\text{N}_4$ in TCN-1. Compared with BCN, the remarkably reduced intensity and widened full width at half maximum (FWHM) of (002) peak in TCN-1 indicates the increased distance of the adjacent $\text{g-C}_3\text{N}_4$ interlayers, suggesting the successful exfoliation of BCN into CN-NS after the ethanol-thermal treatment at 160°C further. As the synthetic temperature increased to 200°C , a new sharp peak corresponded to the melem structure (shown in Fig. 5(c)) is appeared at 23.8° in TCN-2, implying that the C-N chemical bonds between the adjacent tri-s-triazine units start to break at 200°C .⁴⁵ Therefore, it can be speculated that the break of C-N chemical bonds between the adjacent tri-s-triazine units leading to the successful exfoliation of BCN into CN-NC. Once the synthetic temperature increased to 240°C , the amplified intensity and narrowed FWHM of (002) peak in $\text{g-C}_3\text{N}_4$ indicates the increased crystallinity of $\text{g-C}_3\text{N}_4$ in TCN-3 heterostructures. Meanwhile, the significantly heightened intensity of the peak at 23.8° illustrates the increased proportion of melem, meaning the further break of C-N chemical bonds between the adjacent tri-s-triazine structures under the enhanced synthetic temperature at 240°C . Consistent with the results of TEM images, the synthetic temperature plays an essential role in determining the structure of $\text{g-C}_3\text{N}_4$ in the fabricated TCN heterostructures. In addition, the FWHM of

(101) peak in anatase TiO_2 decreases with the increases of the synthetic temperature, illustrating the enhanced crystallinity of TiO_2 . And calculated by the Scherrer equation (shown in ESI S11†), the size of (101) facets in the prepared TCN-1, TCN-2 and TCN-3 are 8.06 nm, 9.07 nm and 10.74 nm, respectively. Therefore, the size of (101) facets in the prepared samples increases with the increase of the synthetic temperature. And this tendency is consistent with the results of TEM images.

FT-IR spectra were used to detect the surface chemical bond of the fabricated TCN heterostructures. As shown in Fig. 5(b), the prominent vibration peak of anatase TiO_2 at $400\text{--}700\text{ cm}^{-1}$ corresponds to the Ti-O-Ti, and the absorption peak at 3500 cm^{-1} denotes the stretching vibration of the hydroxyl ($-\text{OH}$) group. For BCN, the peak at 809 cm^{-1} could be indexed to the typical bending mode of triazine units. The broad absorption band ranging from 1238 to 1638 cm^{-1} is assigned to the characteristic stretching vibration of C-N and C=N heterocycles. The band recorded between 3000 and 3500 cm^{-1} is ascribed to the N-H stretching vibration originated from the bridged or terminal amino group.⁴⁶ Consistent with the results of XRD patterns, the well-reserved characteristic peaks of $\text{g-C}_3\text{N}_4$ indicate that the core chemical structure of $\text{g-C}_3\text{N}_4$ remains stable after the ethanol-thermal treatment. It is deserved to mention that the intensity of the characteristic vibration peaks

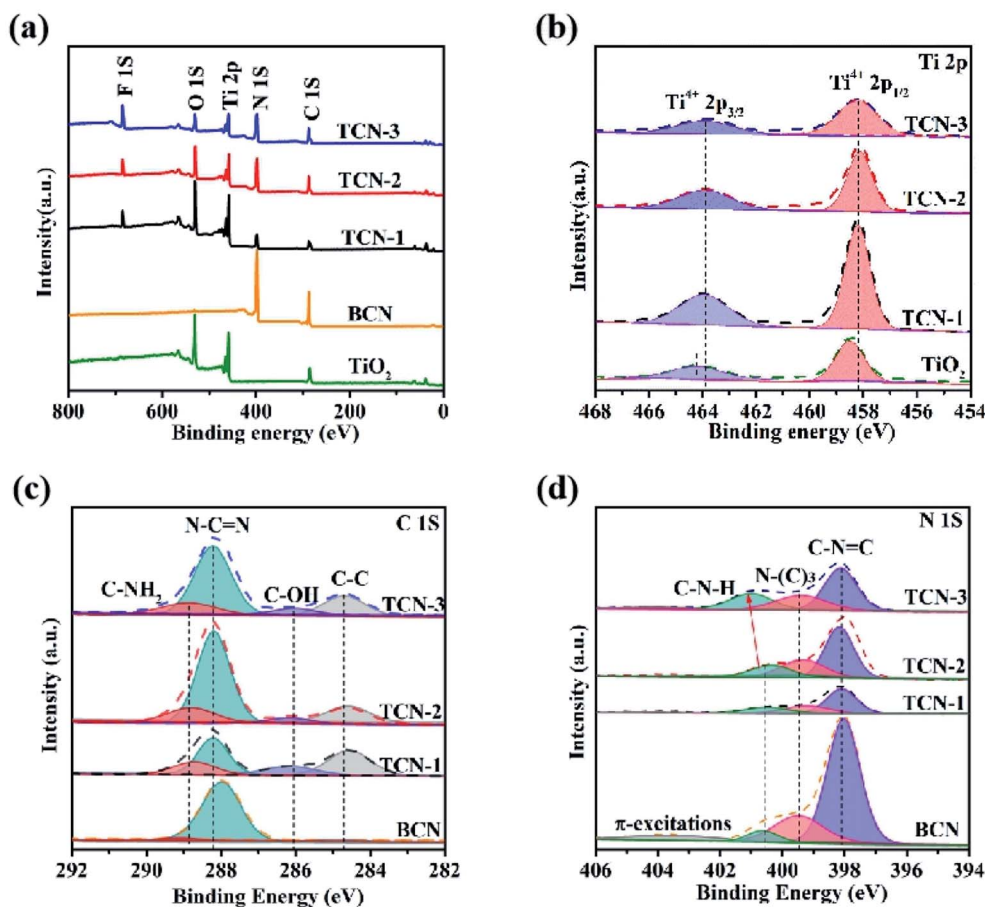


Fig. 6 (a) The survey XPS spectra, the high-resolution XPS of (b) Ti 2p, (c) C 1s and (d) N 1s spectra of BCN, TiO_2 TCN-1, TCN-2 and TCN-3.

between 3000 cm^{-1} and 3500 cm^{-1} increases with the increasing of the synthetic temperature (shown in Fig. S5†). The increased exposure of N-H group could be attributed to the increased break extent of C-N chemical bonds between the adjacent tri-s-triazine units under the enhanced synthetic temperature. Meanwhile, the -OH group of ethanol molecule would nucleophilic attack on the amino group under the high synthetic temperature, thus introducing the -OH group on the surface of $\text{g-C}_3\text{N}_4$.⁵⁸ The introduced -OH group could increase the surface hydrophilicity of the fabricated TCN heterostructures, which is helpful for enhancing the photocatalytic activity of H_2 evolution.⁴⁷

The surface chemical states of the samples were clarified by XPS further. As shown in Fig. 6(a), the survey XPS spectra confirms the coexistence of C, N, Ti and O elements in the prepared TCN hetero-structures. In the high-resolution Ti 2p spectra (shown in Fig. 6(b)), the two dominant peaks at 458.2 eV and 464.1 eV could be assigned to the Ti $2\text{p}_{3/2}$ and Ti $2\text{p}_{1/2}$ energies of Ti^{4+} in TiO_2 , respectively. Compared with pristine TiO_2 , the peak position of Ti 2p in the TCN heterostructures shift negatively (0.6 V), which is induced by the changed electron concentration. In theory, the electron negativity of N element is relatively weaker than O element. Therefore, the negatively shifted peak position of Ti 2p could be attributed to the replacement of O element by the N element, verifying the formation of Ti-N chemical bond. The intimate contact of TiO_2 and $\text{g-C}_3\text{N}_4$ could provide an efficient charge transfer pathway and accelerate the photo-generated e^- - h^+ separation by forming the built-in electric field.⁵² In the high-resolution C 1s spectra (shown in Fig. 6(c)), the peaks can be divided into four main features at 284.6, 286.1, 288.2 and 288.9 eV, attributing to the surface adventitious C, C=N bindings, C-OH species and sp^3 -hybridized C in $(\text{N})_2\text{-C=N}$ bindings, respectively.⁴⁹ Compared with BCN, the newly introduced C-OH group in the fabricated TCN heterostructures confirms an effective surface modification of $\text{g-C}_3\text{N}_4$ in the ethanol solution further, this is consistent with the results of FT-IR spectra. The high-

resolution N 1s spectra (shown in Fig. 6(d)) could be split into four peaks at 398.4, 399.7, 401.3 and 404.2 eV, attributing to the N element in the C-N=C groups, N-(C)₃ groups together with the $(\text{N})_2\text{-C=N}$, C-N-H groups and π -excitations, respectively. It can be underlined that the intensity of C-N-H peak in the high-resolution N 1s spectra increases with the increasing of the synthetic temperature. The increased intensity of C-N-H peak demonstrates that the enhanced synthetic temperature could promote the break of C-N bonds between the adjacent tri-s-triazine units under the shear effects of HF acid, this is also consistent with the results of XRD.

Based on the above analyses, the synthetic temperature plays a crucial role in exfoliating the BCN into CN-NS, CN-NC and CN-QD in the fabricated TCN heterostructures with the shearing effects of HF acid (Fig. 7). It has been verified that BCN could be exfoliated into CN-NS when the enthalpy of mixing is minimized.^{48,49} When the layered BCN was immersed in ethanol solvent, the ethanol molecules would conquer the interlayer van der Waals force and embed into the interlayer gallery of $\text{g-C}_3\text{N}_4$ under the assistant of ultrasonic due to the similar surface energy of ethanol to that of CN-NS.⁵¹ Yang *et al.* have been exfoliated BCN into the free-standing CN-NS in the isopropanol solvent.⁵⁰ Then, the successively added TBOT molecules would anchor onto the surface of $\text{g-C}_3\text{N}_4$ due to the formation of coordinated bonds between Ti atoms and N atoms. Afterwards, the TBOT would be *in situ* hydrated to anatase $\text{TiO}_2\{001\}$ on the surface of CN-NS after the solvothermal process with the effects of HF acid. In addition, the *in situ* hydrated anatase $\text{TiO}_2\{001\}$ are favorable for preventing the aggregation of exfoliated CN-NSs caused by the slow solvent evaporation. Meanwhile, the -OH groups would be introduced onto the surface of $\text{g-C}_3\text{N}_4$ due to the strong polar of ethanol solvent under the high synthetic temperature. Therefore, TCN-1 could be synthesized after the ethanol-thermal process at 160°C for 24 h. It has been revealed that the added HF acid could promote the break of C-N chemical bonds between the adjacent tri-s-triazine units under the enhanced synthetic temperature. Thus, the BCN could be

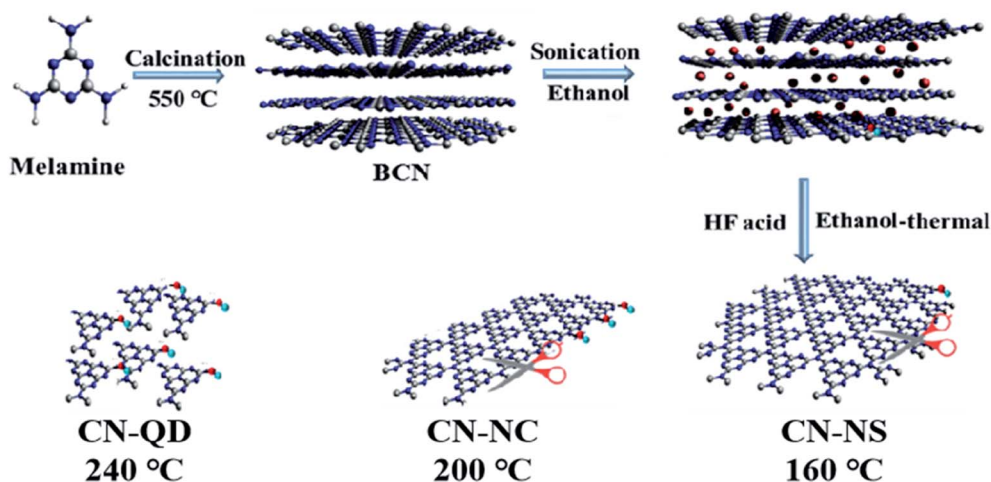


Fig. 7 Schematic illustration of the exfoliation mechanism of BCN into CN-NS, CN-NC and CN-QD.



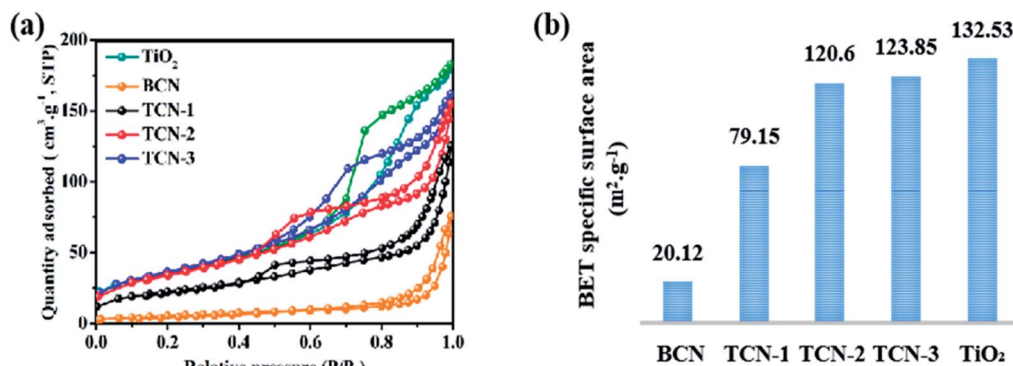


Fig. 8 (a) N₂ Adsorption-desorption isotherms and (b) the surface area values (calculated by the BET equation) of BCN, TCN-1, TCN-2, TCN-3 and TiO₂.

exfoliated into the CN-NC and CN-QD as the heating temperature increased to 200 °C and 240 °C, respectively. Meanwhile, the crystallinity of the prepared TCN increases with the increasing of the synthetic temperature. Therefore, the strongly coupled TCN-1, TCN-2 and TCN-3 were fabricated at 160 °C, 200 °C and 240 °C, respectively.

3.3. Enhancement mechanism of photocatalytic activity

3.3.1 The specific surface area. As presented in Fig. 8(a), anatase TiO₂ exhibits a type-IV isotherm with H₂ hysteresis loops, corresponding to the ink-bottle pores formed between aggregated TiO₂. The hysteresis loop of BCN is the type-II

isotherm with H₃ hysteresis loops, which is the characteristic of slit-like pores formed between CN-NS.⁵² It can be clearly observed that all the hysteresis loops of TCN-1, TCN-2 and TCN-3 are the overlap of type H₂ and H₃ loops, implying the co-presence of anatase TiO₂ and g-C₃N₄ in the prepared TCN heterostructures. Calculated by the BET equation, the surface area values of TiO₂, BCN, TCN-1, TCN-2 and TCN-3 are characterized to be 132.53, 20.12, 79.15, 120.60, and 123.85 m² g⁻¹ (shown in Fig. 8(b)). The BET specific surface area of TCN heterostructures gradually increases with the increasing of the synthetic temperature, confirming the successful exfoliation of BCN into CN-NS, CN-NC and CN-QD further. The larger BET surface area

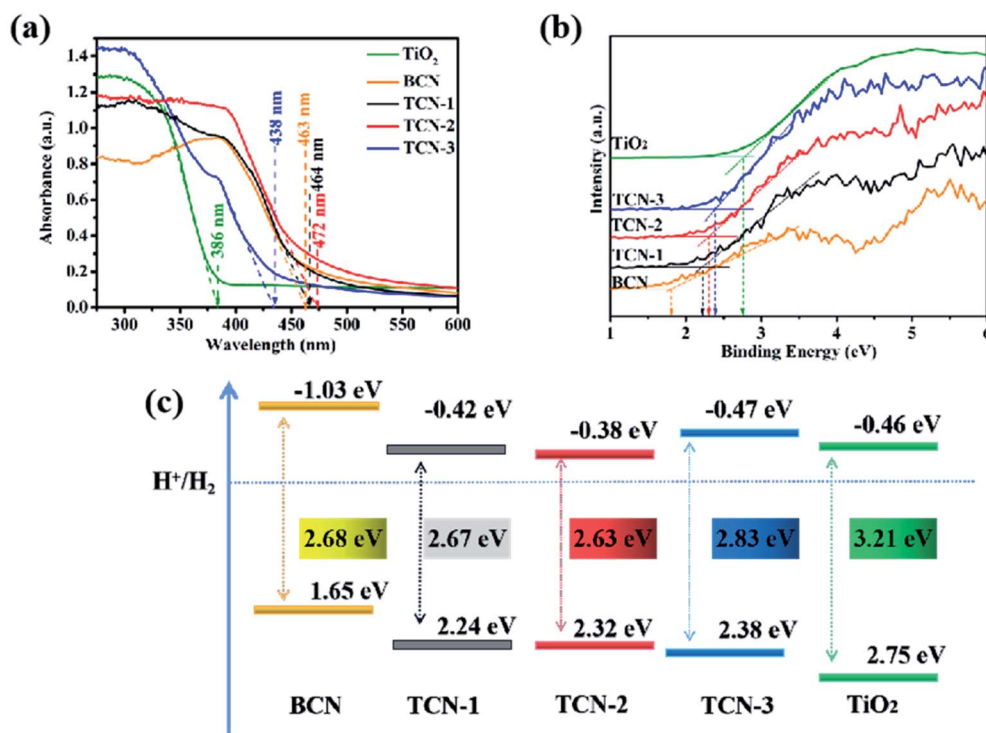


Fig. 9 (a) UV-vis diffuse reflectance spectra, (b) valence band XPS spectra and (c) the corresponding band structure of TiO₂, BCN, TCN-1, TCN-2 and TCN-3.

could result in more reaction active sites for the surface reactions. And it could be concluded that a structural evolution and an increase of surface area can be induced by enhancing the temperature of ethanol-thermal reaction.

3.3.2 Optical properties. UV-vis DRS were detected to explore the optical properties and electronic band structure of the samples. It is obvious in Fig. 9(a) that the absorption intensity increases with the increasing of the synthetic temperature. The enhanced optical absorption could increase the number of incident photons, which is also important for improving photocatalytic activity of H_2 evolution. It has been verified that the light-harvesting ability is closely related to the structural rigidity determined by the crystallinity of the samples.⁵² Therefore, the crystallinity of the samples increases with the increasing of synthetic temperature, this is consistent with the results of TEM and XRD. Furthermore, the light absorption band of anatase TiO_2 is limited in the UV light region with a wavelength below 387 nm. Due to the intrinsic absorption of the introduced CN-NS, TCN-1 has a strong absorption band in the UV-visible region with a wavelength below 464 nm.⁵³ Compared with TCN-1, a slight bathochromic shift of the absorption edge has happened in TCN-2. It could be attributed to the successful exfoliation of BCN into CN-NC.⁵⁴ On the contrary, the absorption edge of TCN-3 shows the obvious blue-shift in contrast to TCN-2.⁵⁵ It is apparent in Fig. 9(a) that there are two obvious absorption edges in TCN-3, this could be attributed to the changed ratio of TiO_2 to $g-C_3N_4$ in TCN-3 (shown in ESI S9†).⁵⁸ And due to the excellent resistance to the high temperature of TiO_2 , the produced two obvious

absorption edges of TCN-3 could also be used to further confirm that the structure of $g-C_3N_4$ has been changed in TCN-3 as the synthetic temperature increased to 240 °C. According to the Kubelka–Munk equation $E_g = 1240/\lambda_g$ (E_g is the gap energy, and λ_g is the light absorption threshold), the band gaps of TiO_2 , BCN, TCN-1, TCN-2 and TCN-3 are calculated to be 3.21, 2.68, 2.67, 2.60 and 2.83 eV, respectively. The narrowed band gap is expected to have the wider light absorption range, resulting in the enhanced photocatalytic performance under visible light.

To study the band structure of the prepared TCN photocatalysts further, the valence band edge (VB) of the samples were detected by VB-XPS. As shown in Fig. 9(b), the VB edge of TiO_2 , $g-C_3N_4$, TCN-1, TCN-2 and TCN-3 occurs at 2.75 eV, 1.65 eV, 2.24 eV, 2.32 eV and 2.38 eV, respectively. According to the band gap measured by UV-vis DRS, their CB are calculated to be −0.46 eV, −1.03 eV, −0.42 eV, −0.38 eV and −0.47 eV, respectively. The corresponding band structure of the prepared samples is shown in Fig. 9(c). The differences in the band gaps are mainly caused by the diverse structure of $g-C_3N_4$ in the prepared TCN photocatalysts. Since the CB of anatase TiO_2 is lower than that of $g-C_3N_4$, there is sufficient Gibbs free energy to induce the electrons to inject from $g-C_3N_4$ into TiO_2 , which is conducive to separating the electron–hole pairs.⁵³

3.3.3 Charge transfer properties. To gain better insight into the charge separation and transfer properties of the samples, the photoluminescence spectra (PL), photocurrent response test and EIS analyses were conducted. The intensity of PL emission spectrum is dominated by the recombination of photo-generated e^-h^+ pairs between the lone pair valence band and

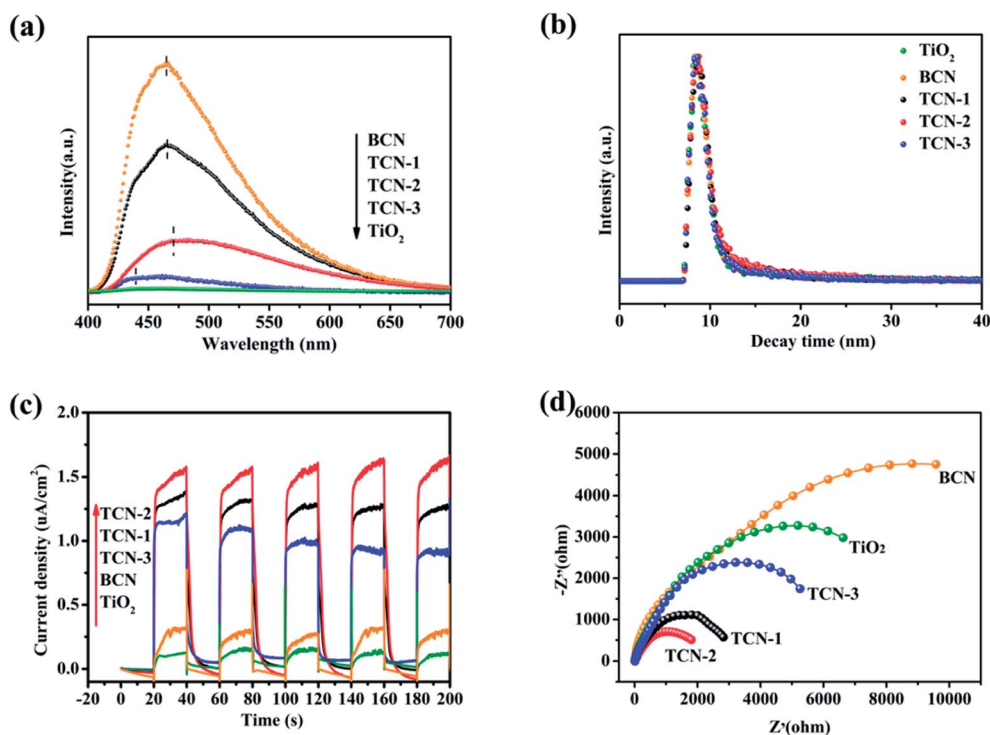


Fig. 10 (a) PL spectra, (b) time-resolved photoluminescence decay spectra ($\lambda_{ex} = 387$ nm), (c) transient photocurrent response and (d) EIS Nyquist plots of TiO_2 , BCN, TCN-1, TCN-2, TCN-3 heterostructure.



π^* conduction band.⁵⁶ The lower intensity of the PL emission spectrum means the lower the recombination possibility of photo-generated e^-h^+ pairs. As shown in Fig. 10(a), the emission peak intensity for TCN-1 decreases rapidly in comparison with BCN, suggesting that the recombination of photo-generated pairs is available suppressed by forming the heterojunction between CN-NS and TiO_2 . It is obvious that the intensity of PL peak gradually reduces with the increasing of the synthetic temperature, suggesting the increased separation efficiency of photo-generated carries. Moreover, the transient fluorescence spectra (shown in Fig. 10(b)) were recorded to directly elucidate the lifetime of photo-generated charge carriers, and the results were summarized in Table S3.[†] Consistent with the result of PL emission spectra, the average lifetime of charge carriers for TCN-2 (7.733 ns) is much higher than that of BCN (5.870 ns), implying that the separation of photo-generated pairs in TCN-2 is significantly facilitated. While it's worth noting that the lowest peak intensity of TCN-3 is induced by the weak photo-response under the visible-light irradiation according to the results of UV-vis DRS.

The transient photocurrent response tests were carried out to investigate the migration efficiency of the photo-generated carriers further. Herein, the increased photocurrent could be majorly ascribed to the efficient photo-generated separation and transferring, which is benefit for the photocatalysis and correspond to the results of photocatalytic performance. As shown in Fig. 10(c), both the pure TiO_2 and BCN exhibit low photocurrent intensity under visible-light irradiation. Obviously, the photocurrent responses of all the TCN

heterostructures are significantly prompted, suggesting the enhanced transfer efficiency of photo-generated carriers at the strongly coupled interface between TiO_2 and $g-C_3N_4$.⁴⁰ In particular, TCN-2 exhibits the highest photocurrent density, implying the best separation ability of photo-generated carriers. In addition, the EIS Nyquist plots obtained in the dark could be used to evaluate the inherent interface resistance of the prepared samples. The smaller impedance arc radius in the Nyquist plot implies a weaker charge transfer resistance, valid mobility, and easier carriage of photo-generated carriers. Therefore, EIS is also deemed as valid method to investigate the migration rates of photo-generated charge carriers.⁵⁰ The smaller impedance arc radius in the Nyquist plot implies a weaker charge transfer resistance, valid mobility, and easier carriage of photo-generated carriers. In Fig. 10(d), the Nyquist plots radius of TCN-2 is smaller than that of BCN, indicating the faster charge transfer and the reduced chemical impedance. And the smallest Nyquist plots radius is also obtained at TCN-2. Combined with the above analyses, the structure modification of $g-C_3N_4$ in the fabricated TCN heterostructures could affect the specific surface areas, the visible-light absorption ability, the recombination and migration rates of the photo-generated charge carriers. It has been proved that the larger specific surface areas, faster migration rates of the photo-generated e^-h^+ pairs and higher visible-light response ability are helpful for achieving the better photocatalytic effect.⁶² Thus, the fabricated TCN-2 is expected to have the best photocatalytic activity of H_2 evolution.

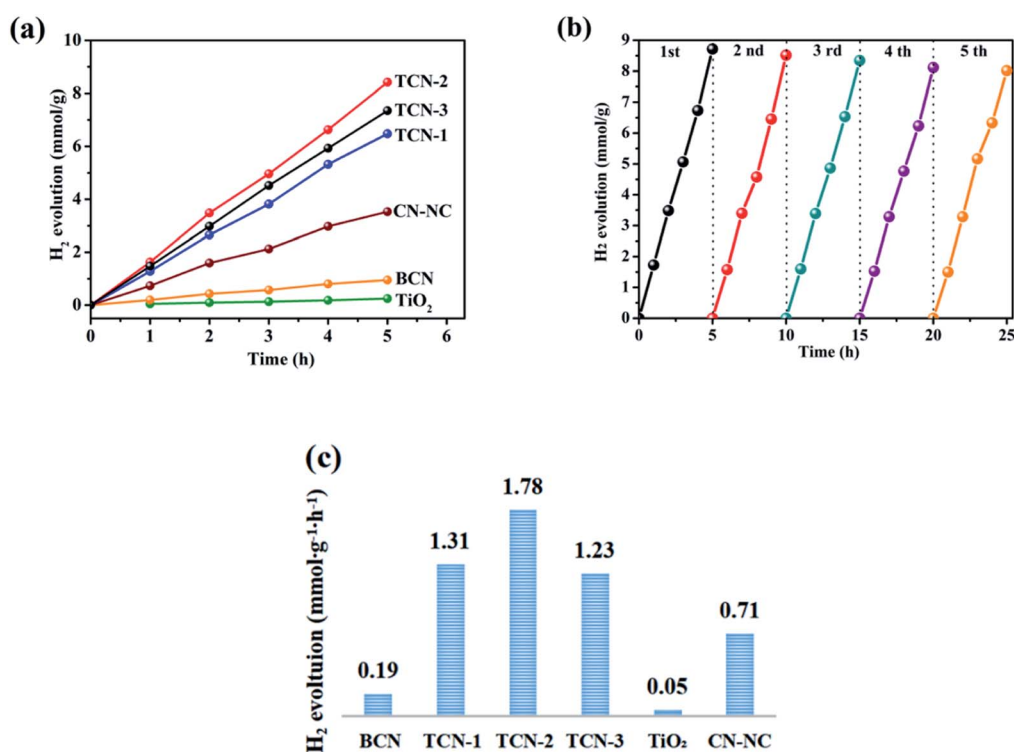


Fig. 11 (a) The photocatalytic H_2 evolution of different samples with the prolong of the time, (b) the recyclability tests under the visible light ($\lambda > 420$ nm) for the TCN-2 and (c) the photocatalytic H_2 evolution rate of different samples.

3.4. Photocatalytic H₂ evolution

The photocatalytic reduction activities of the samples were revealed by water splitting into H₂ under visible light irradiation. As demonstrated in Fig. 11(a) and (c), the anatase TiO₂{001} exhibits the weaker H₂ production of 0.05 mmol g⁻¹ h⁻¹ since its bandgap energy is too wide for visible light to induce electron excitation.⁶¹ Suffering from fast recombination of photoinduced carriers and small surface area, the pure BCN also shows a low H₂ evolution of 0.19 mmol g⁻¹ h⁻¹. The photocatalytic performance of H₂ evolution for TCN-1, TCN-2 and TCN-3 exhibits the tendency of increasing first and then decreasing, and gets an optimal value at TCN-2. The prepared TCN-2 shows conspicuous H₂ evolution rates as high as 1.78 mmol g⁻¹ h⁻¹, which is about 9.37 times higher than BCN and 35.6 times higher than TiO₂. It is obvious that the H₂ evolution performance of the prepared samples exhibits a noticeable enhancement after cooperating anatase TiO₂{001} with g-C₃N₄. To further verify the vital effects of the formed heterojunction between TiO₂{001} and g-C₃N₄ on enhancing the photocatalytic activity of H₂ evolution, the photocatalytic performance of H₂ evolution for the pure CN-NC was also measured. The prepared pure CN-NC shows a lower H₂ evolution of 0.71 mmol g⁻¹ h⁻¹ as compared with that of TCN-2. It indicates that the formation of covalent bonding at TCN heterostructure would be the main factor for the photocatalytic H₂ evolution enhancement. And the synthetic temperature of the prepared samples has close interaction with the H₂ evolution activity by affecting the structure of g-C₃N₄ and the crystallinity

of TiO₂{001}. And the exfoliated CN-NC in TCN-2 could provide more surface sites, enhance the light absorption ability and fast the mass transfer efficiency. And the increased crystallinity of TiO₂{001} is also helpful in improving the photocatalytic activity of H₂ evolution.⁶³ Through a detailed comparison of the photocatalytic H₂ evolution activity of the prepared TCN with that of the photocatalysts prepared by other groups (shown in ESI S8†), the prepared TCN-2 also exhibits an excellent activity of H₂ evolution among the g-C₃N₄/TiO₂ systems. Based on the remarkable photocatalytic H₂ evolution performance, it's significant to explore the cycling stability of TCN-2. As shown in Fig. 11(b), the rate of photocatalytic H₂ production has no apparent loss after five cycling experiments, indicating that the prepared TCN-2 is stable and difficult to decompose during the process.

Combined with DFT calculation, the mechanism of the enhanced photocatalytic activity for TCN-2 was proposed at atom scale. It has been reported that the conduction band minimum (CBM) and valence band maximum (VBM) states of TiO₂ are mainly contributed by Ti 3d states and O 2p states, respectively.⁶⁰ According to the calculated TDOS and the corresponding PDOS of TCN-2 shown in Fig. 12(b), the CBM and the VBM of the TCN-2 are mainly composed of the Ti 3d states in TiO₂{001} and the N 2p states in g-C₃N₄, respectively, confirming the formation of intimate contact between TiO₂ and g-C₃N₄ in TCN-2 further. As shown in Fig. 9(c), the VB potential positions of BCN and TiO₂{001} are 1.65 eV and 2.75 eV (vs. NHE), and their CB are -1.03 eV and -0.46 eV (vs. NHE), respectively.⁵⁹

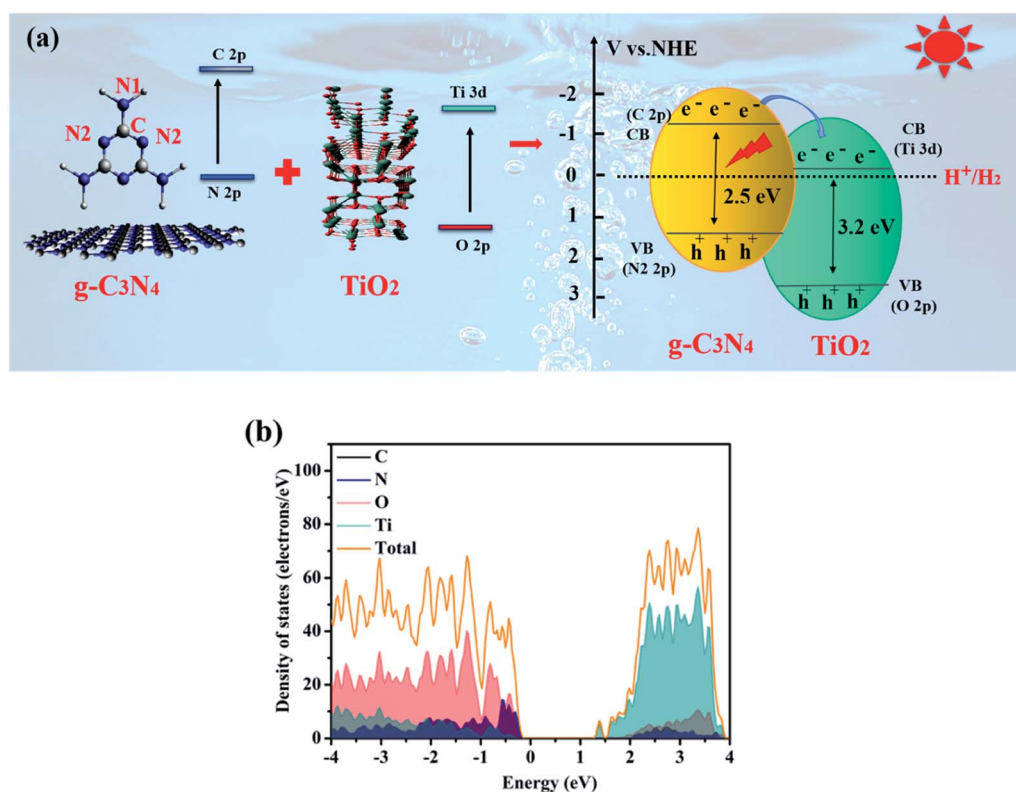


Fig. 12 (a) Proposed photocatalytic reaction mechanism of TCN-2 heterostructure, (b) calculated TDOS and the corresponding PDOS of TCN-2.



And the CB potential position of TCN-2 is -0.38 eV (vs. NHE), which is similar to that of TiO_2 , confirming the formation of type II heterojunction between $\text{TiO}_2\{001\}$ and $\text{g-C}_3\text{N}_4$ in TCN-2 (shown in Fig. 12(a)). Correspondingly, the VB potential position of TCN-2 is 2.32 eV (vs. NHE). For TCN-2, the e^- - h^+ pairs could be generated in $\text{g-C}_3\text{N}_4$ under visible light illumination ($\lambda > 420$ nm). While the CB offset between $\text{g-C}_3\text{N}_4$ and TiO_2 will prompt the photo-generated electrons transfer from the CB of $\text{g-C}_3\text{N}_4$ to the CB of TiO_2 . It is worth mentioning that the high exposed $\{001\}$ facets of anatase TiO_2 and the heterojunction structure between $\text{TiO}_2\{001\}$ and $\text{g-C}_3\text{N}_4$ could inhibit the recombination of photoinduced charge carriers. In addition, the enlarged specific surface area could provide more reactive sites and the introduced $-\text{OH}$ group is also favourable for the adsorption of H_2O molecules, those are beneficial for enhancing the photocatalytic activity of TCN-2.

4. Conclusions

In conclusion, we presented a facial *in situ* ethanol-thermal method for synthesizing the high-crystalline $\text{TiO}_2/\text{g-C}_3\text{N}_4$ (TCN) heterostructures with intimate contact. The structure of $\text{g-C}_3\text{N}_4$ in the fabricated TCN heterostructures could be modulated from BCN to CN-NS, CN-NC or CN-QD by regulating the synthetic temperature. Through the detailed characterizations, the structure evolution mechanism of $\text{g-C}_3\text{N}_4$ was ascribed to the co-effect of the ethanol-thermal treatment and the added HF acid with shearing effect. As expected, the fabricated TCN-2 shows the highest efficient photocatalytic activity under visible light irradiation, and the H_2 production rate reaches up to $1.78 \text{ mmol h}^{-1} \text{ g}^{-1}$, which is 35.6 times higher than that anatase TiO_2 . Confirmed by DFT simulation, the excellent photocatalytic H_2 evolution performance could be attributed to not only the synergistic effect of the CN-NCs with enhanced visible absorption and anatase $\text{TiO}_2\{001\}$, but also the formed compact contact interface and the introduced surface adsorbed $-\text{OH}$ group during *in situ* ethanol-thermal process. This work opens up a new window to the fabrication of highly active strongly coupled TCN heterostructures with the desired structure for the efficient photocatalytic H_2 evolution under visible-light irradiation.

Conflicts of interest

The authors solemnly declare no competing financial interests.

Acknowledgements

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References

- X. B. Chen, S. Shen, L. Guo and S. S. Mao, *Chem. Rev.*, 2010, **110**, 6503–6570.
- K. Maeda, K. Teramura, D. Lu, T. Takata, N. Saito, Y. Inoue and K. Domen, *Nature*, 2006, **440**, 295–298.
- R. T. Asahi, T. Morikawa, T. Ohwaki, K. Aoki and Y. Taga, *Science*, 2001, **293**, 269–271.
- A. Fujishima and K. I. Honda, *J. Soc. Chem. Ind., Jpn.*, 1971, **74**, 355–358.
- J. Schneid, M. Matsuoka, M. Takeuchi, J. Zhang, M. Anpo and D. W. Bahnemann, *Chem. Rev.*, 2014, **14**, 9919–9986.
- Y. Yang, Q. Z. Deng and Y. L. Zhang, *Chem. Eng. J.*, 2019, **360**, 1247–1254.
- H. G. Yang, C. H. Sun, S. Z. Qiao, J. Zou and G. Liu, Anatase TiO_2 single crystals with a large percentage of reactive facets, *Nature*, 2008, **453**, 638–643.
- H. G. Yang, G. Liu, S. Z. Zhang, C. H. Sun, Y. G. Jin, S. C. Smith, J. Zou, H. M. Chen and G. Q. Lu, *J. Am. Chem. Soc.*, 2009, **131**, 4078–4083.
- H. Zhang, X. J. Lv, Y. M. Li, Y. Wang and J. H. Li, *ACS Nano*, 2010, **4**, 380–386.
- A. Naldoni, M. Altomare, G. Zoppellaro, N. Liu, S. Kment, R. Zboril and P. Schmuk, *ACS Catal.*, 2019, **9**, 345–364.
- T. M. Su, Q. Shao, Z. Z. Qin, Z. H. Guo and Z. L. Wu, *ACS Catal.*, 2018, **8**, 2253–2276.
- Y. F. Chen, W. X. Huang, D. L. He, Y. Situ and H. Huang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 14405–14414.
- S. W. Cao, J. X. Low, J. G. Yu and M. Jaroniec, *Adv. Mater.*, 2015, **27**, 2150–2176.
- X. Wang, S. Blechert and M. Antonietti, *ACS Catal.*, 2012, **2**, 1596–1606.
- W. Wang, J. Fang, S. Shao, L. Min and C. Lu, *Appl. Catal., B*, 2012, **217**, 57–64.
- Y. Tan, Z. Shu, J. Zhou, T. T. Li, W. B. Wang and Z. L. Zhao, *Appl. Catal., B*, 2018, **230**, 260–268.
- Q. Wang, L. Zhang, Y. Guo, M. Shen and J. Shi, *Chem. Eng. J.*, 2020, **396**, 125347–125359.
- F. Guo, H. Sun, X. Huang, W. Shi and C. Yan, *J. Chem. Technol. Biotechnol.*, 2020, **95**, 2684–2693.
- G. Li, Z. Lian, W. Wang, D. Zhang and H. Li, *Nano Energy*, 2016, **19**, 446–454.
- Y. Q. Sheng, Z. Wei, H. Miao, W. Q. Yao, H. Q. Li and Y. F. Zhu, *Chem. Eng. J.*, 2019, **370**, 287–294.
- B. Zhang, X. He, X. H. Ma, Q. H. Chen and Y. Xin, *Sep. Purif. Technol.*, 2020, **247**, 116932–116941.
- Y. W. Zhang, J. S. Xu, J. Mei, S. Sarina, Z. Y. Wu, T. Liao, C. Yan and Z. Q. Sun, *J. Hazard. Mater.*, 2020, **394**, 122529–122536.
- J. Wang, G. H. Wang, X. Wang, Y. Wu, Y. R. Su and H. Tang, *Carbon*, 2019, **149**, 618–626.
- G. S. Li, Z. C. Lian, W. C. Wang, D. Zhang and H. Li, *Nano Energy*, 2016, **19**, 446–454.
- B. Y. Shao, J. Y. Wang, Y. Z. Zhang, X. Tan, W. Zhou and Y. L. Chen, *Catal. Sci. Technol.*, 2021, **11**, 1282–1291.
- M. Y. Liu, K. Jiang, X. Ding, S. L. Wang, C. Zhang, J. Liu, Z. Zhan, G. Cheng, B. Li and H. Chen, *Adv. Mater.*, 2019, **31**, 1807865–1807871.
- L. H. Lin, Z. Y. Yu and X. C. Wang, *Angew. Chem., Int. Ed.*, 2019, **58**, 6164–6175.



- 28 W. Iqbal, B. C. Qiu, Q. H. Zhu, M. Xing and J. Zhang, *Appl. Catal., B*, 2018, **232**, 306–313.
- 29 K. Schwinghammer, M. B. Mesch, V. Duppel, C. Ziegler, J. Senker and B. V. Lotsch, *J. Am. Chem. Soc.*, 2014, **136**, 1730–1732.
- 30 P. F. Xia, M. Antonietti, B. C. Zhu, T. Heil and J. Yu, *Adv. Funct. Mater.*, 2019, **29**, 1900093–1900101.
- 31 Y. Li, D. Zhang, X. Feng and Q. Xiang, *Chin. J. Catal.*, 2020, **41**, 21–30.
- 32 U. Caudillo-Flores, M. J. Muñoz-Batista, R. Luque, M. Fernández-García and A. Kubacka, *Chem. Eng. J.*, 2019, **378**, 122228–122239.
- 33 K. Hu, R. Li, C. Ye, A. Wang, W. Q. Wei, D. Hu, R. Qiu and K. Yan, *J. Cleaner Prod.*, 2020, **253**, 120055–120063.
- 34 S. H. Liu and W. X. Lin, *J. Hazard. Mater.*, 2019, **368**, 468–476.
- 35 R. Acharya and K. Parida, *Chem. Eng.*, 2020, **8**, 103896–103918.
- 36 S. W. Cao, J. X. Low, J. G. Yu and M. Jaroniec, *Adv. Mater.*, 2015, **27**, 2150–2176.
- 37 Q. Shang, X. Tan, T. Yu, Z. Zhang, Y. Zou and S. Wang, *J. Colloid Interface Sci.*, 2015, **455**, 134–144.
- 38 J. Xu, L. Zhang, R. Shi and Y. Zhu, *J. Mater. Chem. A*, 2013, **1**, 14766–14772.
- 39 J. S. Zhang, Y. Chen and X. C. Wang, *Energy Environ. Sci.*, 2015, **8**, 3092–3108.
- 40 K. Li, S. Gao, Q. Wang, Z. Huang and B. Ying, *ACS Appl. Mater. Interfaces*, 2015, **7**, 9023–9030.
- 41 T. Xiong, W. Cen, Y. Zhang and F. Dong, *ACS Catal.*, 2016, **6**, 2462–2472.
- 42 Q. Guo, C. Zhou, Z. Ma and X. Yang, *Adv. Mater.*, 2019, **31**, 1901997.
- 43 Y. Y. Li, B. X. Zhou, H. W. Zhang, S. F. Ma, W. Q. Huang, W. Peng, W. Hu and G. F. Huang, *Nanoscale*, 2019, **11**, 6876–6885.
- 44 W. Wang, J. Yu, Z. Shen, D. L. Chan and T. Gu, *Chem. Commun.*, 2014, **50**, 10148–10150.
- 45 Y. X. Li, S. Ouyang, H. Xu, W. Hou, M. Zhao, H. Chen and J. Ye, *Adv. Funct. Mater.*, 2019, **29**, 1901024–1901032.
- 46 Q. Yang, C. Chen, Q. Zhang, Z. Zhang and X. Fang, *Carbon*, 2020, **164**, 337–348.
- 47 T. Yu, T. Xie, W. Zhou, Y. Z. Zhang, Y. L. Chen, B. Y. Shao, W. Q. Guo and X. Tan, *ACS Sustainable Chem. Eng.*, 2021, **9**, 7529–7540.
- 48 D. Lu, P. Fang, W. Wu, J. Ding, L. Jiang, X. Zhao, C. Li, M. Yang, Y. Li and D. Wang, *Nanoscale*, 2017, **9**, 3231–3245.
- 49 S. Huang, Y. Xu, F. Ge, D. Tian, X. Zhu, M. Xie, H. Xu and H. Li, *J. Colloid Interface Sci.*, 2019, **556**, 324–334.
- 50 L. Cheng, Y. Song, H. Chen, G. Liu and W. Jin, *Sep. Purif. Technol.*, 2020, **250**, 117200–117211.
- 51 X. She, J. Wu, J. Zhong, H. Xu, Y. Yang, R. Vajtai, J. Lou, Y. Liu, D. Du and H. Li, *Nano Energy*, 2016, **27**, 138–146.
- 52 P. Xia, B. Zhu, J. Yu, S. Cao and M. Jaroniec, *J. Mater. Chem. A*, 2017, **5**, 3230–3238.
- 53 X. Gao, J. Feng, D. Su, Y. Ma, G. Wang, H. Ma and J. Zhang, *Nano Energy*, 2019, **59**, 598–609.
- 54 Z. Liu, G. Wang, H. S. Chen and Y. Ping, *Chem. Commun.*, 2018, **54**, 4720–4723.
- 55 X. Zhang, H. Wang, H. Wang, Q. Zhang, J. Xie, Y. Tian and Y. Xie, *Adv. Mater.*, 2014, **26**, 4438–4443.
- 56 Q. Han, B. Wang, J. Gao, Z. Cheng, Y. Zhao, Z. Zhang and L. Qu, *ACS Nano*, 2016, **10**, 2745–2751.
- 57 J. Wang, H. Jie, H. Xie and A. Qu, *Int. J. Hydrogen Energy*, 2014, **39**, 6354–6363.
- 58 L. J. Wang, G. Zhou, Y. Tian, L. Yan, M. Deng, B. Yang, Z. Kang and H. Sun, *Appl. Catal., B*, 2019, **244**, 262–271.
- 59 L. N. Kong, X. T. Zhang, C. H. Wang, J. P. Xu, X. W. Du and L. Li, *Appl. Surf. Sci.*, 2018, **448**, 288–296.
- 60 N. Guo, Y. Zeng, H. Y. Li, X. J. Xu, H. W. Yu and X. R. Han, *J. Hazard. Mater.*, 2018, **353**, 80–88.
- 61 Y. Li, X. H. Feng, Z. X. Lu, H. Yin, F. Liu and Q. J. Xiang, *J. Colloid Interface Sci.*, 2018, **513**, 866–876.
- 62 J. Wang, T. Bo, B. Shao, Y. Z. Zhang and L. X. Jia, *Appl. Catal., B*, 2021, **297**, 120498.
- 63 Y. Z. Zhang, W. Zhou, L. X. Jia, X. Tan and Y. L. Chen, *Appl. Catal., B*, 2020, **27**, 119152.

