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## PAPER

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## Tunable formaldehyde sensing properties of palladium cluster decorated graphene\*

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The ability to tune the adsorption strength of the targeted gas on sensing materials is crucial for sensing applications. By employing first-principles calculations the adsorption and sensing properties of HCHO on small Pd<sub>n</sub> (n = 1-6) cluster decorated graphene have been systematically investigated. The adsorption energy is found to depend on the size of the  $Pd_n$  cluster and can be tuned in a wide range from -0.68 eV on Pd(111) to -1.98 eV on the Pd<sub>3</sub>/graphene system. We also find that the Pd<sub>9</sub>/graphene (n = 5 and 6) systems have an appropriate adsorption energy for HCHO gas sensing. The currentvoltage curves are calculated by the non-equilibrium Green's function method for the two-probe nanosensor devices along both the armchair and zigzag directions. The devices constructed with  $Pd_p/$ graphene (n = 5 and 6), having the highest absolute response over 20% at small voltages, should be applicable for HCHO detection. This work provides a theoretical basis for exploring potential applications of metal cluster decorated graphene for gas sensing.

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#### 1. Introduction

Formaldehyde is one of the volatile organic compounds (VOCs) that is chronically and slowly emitted and evaporated from upholstery, carpeting, adhesives, wood and plate products, furniture, and lacquer which are widely used in house construction and decoration.<sup>1,2</sup> Moreover, it is an essential ingredient of consumer goods, such as shampoos, soap, and detergents, and utilized in pharmacology and medicine because of its disinfection effect. Exposure in the circumstances of formaldehyde has been confirmed as a crucial cause of sick building syndrome (SBS) to humans and animals.<sup>2,3</sup> The most common health detriment from formaldehyde is an irritation of the eyes, nose, throat, and skin mucous membrane, and with the increase in the formaldehyde concentration, nausea, cough, headache and other symptoms can appear. Under conditions of unusually high or prolonged exposure, formaldehyde is categorized as a mutagen and possible human carcinogen and teratogens.4-6 Therefore, efficient and convenient methods to detect formaldehyde have been strongly demanded for indoor and outdoor environmental measurement and control. The development of formaldehyde sensors is thought to be

a desirable means for the gas monitoring because the gas sensors own the advantages of excellent selectivity, fast response, low time-consumption, simple operation, and realtime detection in situ condition.7-9

The graphene decorated by metal nanostructures has become promising materials for various applications, such as chemical sensors, electronic devices, energy storage and fuel cells.10,11 These materials have caught great scientific and technological interests.12-15 Recently, the graphene sheets functionalized with metal nanoparticles had shown to increase the selectivity, sensitivity, limit of detection, response, or an unio of these performances.<sup>16-19</sup> Gutés et al.<sup>16</sup> exploited an simple and easy-to-implement method of electroless deposition of Au nanoparticles to fabricate Au-decorated graphene sensor for H<sub>2</sub>S detecting in relatively fast response. Dong et al.<sup>17</sup> developed a biosensor synthesized by assembling Au-Pd/ graphene nanocomposites with high electroactive surface and electrical conductivity, which resulted in sensing toward H<sub>2</sub>O<sub>2</sub> in cancer cell with a low detection limit, a good linear range, and a rapid response. Kumar et al.<sup>18</sup> reported the fast response and recovery of H2 sensing of Pd-Pt alloy nanoparticle-graphene composite layers fabricated using chemical methods. Fan et al.<sup>19</sup> found that Au nanoparticles decorated graphene could enhanced the NO<sub>2</sub> detecting with fast response, good reversibility and mechanical stability, and high sensitivity. It had demonstrated that palladium nanoparticles could be the very effective oxidation catalysts and electrochemical sensors for formaldehyde.<sup>12,13,20-23</sup> The formaldehyde detecting properties of single-walled carbon nanotube hybridized with Pd were investigated by density-functional theory calculations.<sup>22</sup> These researches indicated that metal nanoparticles decorated

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graphene shows a great potential for applying to the gas sensing and catalysis.

In this paper, we investigate the HCHO absorption and sensing properties of  $Pd_n$  (n = 1-6) clusters supported on graphene by using non-equilibrium Green's function (NEGF) method combined with the density-functional theory (DFT). We find that it is possible to adjust the adsorption strength of HCHO gas on  $Pd_n$  (n = 1-6) clusters supported on graphene by changing the size of Pd<sub>n</sub> cluster. As increasing the size of Pd cluster the adsorption strength decreases and the adsorption energy approaches the value of -0.68 eV for the HCHO adsorption on Pd(111), which is beyond the low limit of targeted gas adsorption energy -0.8 eV required by good sensitivity.24 The geometries of  $Pd_n$  nanoclusters supported on graphene for the both cases with and without HCHO adsorption are optimized, and the HCHO adsorption energies are calculated. The two-probe nano-sensor devices consisted of the left/right electrodes and the central scattering region are constructed, and the current-voltage curves and sensor responses are simulated. The results indicate that the systems of  $Pd_n$  (n = 5 and 6) clusters supported on graphene having the proper adsorption energy and a large enough response at low bias voltages are promising for HCHO sensing applications.

### 2. Computational details

All the structural optimizations and electronic property calculations are performed by using the Vienna Ab initio Simulation Package (VASP),<sup>25,26</sup> adopting the generalized gradient approximation (GGA) in the Perdew, Burke and Ernzernhof (PBE) functional27 for exchange-correlation potential and the projector augmented wave (PAW) method<sup>28</sup> to describe the electron-ion interaction. A conjugate-gradient (CG) algorithm is performed to relax all the ions until the force acting on each ion is less than 0.01 eV Å<sup>-1</sup>. The graphene is modelled by a  $(4 \times 4)$ supercell containing 32 carbon atoms and a 15 Å thick vacuum in the z direction which is perpendicular to the graphene surface. A large  $(5 \times 5)$  supercell is also used for testing the lateral interaction imposed by the periodic boundary conditions and we find that the lateral interaction in the  $(4 \times 4)$  graphene supercell model is negligible. The vacuum thickness is large enough to avoid interactions with the adjacent cells due to the periodic boundary conditions enforced in the calculations. The following electronic states are chosen as the valence electron configurations in the pseudopotential description of the system: 1s for H, 2s2p for C and O, and 4d5s for Pd. A semiempirical DFT-D2 (ref. 29) correction is employed to describe van der Waals interactions between the graphene substrate and gas molecule. The more sophisticated D3 correction method has also tested in our calculations and the results show that the HCHO adsorptions on Pd<sub>n</sub>/graphene become slightly weaker by about 0.1 eV. The plane-wave cutoff energy is set to be 500 eV, and a Gaussian smearing for the occupation of the electronic levels has been adopted in all DFT simulations. The Brillouin zone integrations are performed with  $5 \times 5 \times 1$  and  $15 \times 15 \times 1$ Monkhorst-Pack meshes for atomic relaxations and electronic structure calculations, respectively. The theoretical C-C bond

length in graphene is 1.43 Å, which is in excellent agreement with the previous work.<sup>30</sup> For HCHO adsorption on the Pd<sub>n</sub>/ graphene substrate, the charge transfer  $\Delta q$  is obtained based on the Bader charge analysis.<sup>31,32</sup> Positive  $\Delta q$  means the electron gain of adsorbed gas molecule from the substrate.

The formaldehyde adsorption energy on the  $Pd_n/graphene$  substrate is calculated as follows:

$$E_{\rm ads} = E_{\rm HCHO-Pd_{\eta}/Gr} - E_{\rm Pd_{\eta}/Gr} - E_{\rm HCHO}$$
(1)

where  $E_{\text{HCHO-Pd}_n/\text{Gr}}$ ,  $E_{\text{Pd}_n/\text{Gr}}$  and  $E_{\text{HCHO}}$  represent the total energies of the system of HCHO molecule adsorbed on Pd<sub>n</sub>/graphene substrate, the Pd<sub>n</sub> cluster loaded on graphene substrate and an isolated HCHO molecule, respectively. A negative  $E_{\text{ads}}$  means that the adsorption is exothermic.

The electronic transport properties are investigated by Nanodcal,<sup>33,34</sup> which employs the non-equilibrium Green's function technique with the Keldysh formalism in combination with DFT.<sup>35,36</sup> The GGA-PBE functional<sup>27</sup> is adopted as the exchange-correlation potential and double  $\zeta$ -plus basis (DZP) is employed for the basis set. The real space mesh cutoff energy is 100 Ha, and the Brillouin zone samplings are represented by Monkhorst–Pack mesh of  $1 \times 15 \times 1$  and  $1 \times 15 \times 100$  *k*-points for the electrode and central region of the two-probe geometry, respectively.

The electronic transmission function through the device is obtained by the Landauer formula,<sup>37</sup> as follows

$$T(E, V_{\rm b}) = T_{\rm r}[\Gamma_{\rm L}(E)G^{\rm R}(E)\Gamma_{\rm R}(E)G^{\rm A}(E)]$$
(2)

where  $G^{\text{R/A}}$  is the retarded/advanced Green's function in the central scattering region, and  $\Gamma_{\text{L/R}}$  is the coupling matrix between the central zone and the left/right leads.<sup>33</sup> The transmission around the Fermi energy at the equilibrium state is  $T(E,V) = T_{\text{r}}[\Gamma_{\text{L}}G^{\text{R}}\Gamma_{\text{R}}G^{\text{A}}]$ , which represents the probability that an electron with the energy *E* transmits from the left electrode into the right one. For non-equilibrium transport properties, the electrochemical potentials  $\mu_{\text{L/R}} = \mu \pm eV_{\text{b}}/2$ , where  $\mu$  and  $V_{\text{b}}$  are the Fermi level and the applied bias, are applied to the left and right electrodes, respectively. The current through the device is given using the Landauer–Büttiker equation,<sup>38</sup>

$$I(V_{\rm b}) = G_0 \int T(E, V_{\rm b}) [f(E - \mu_{\rm L}) - f(E - \mu_{\rm R})] dE$$
(3)

where the  $G_0 = \frac{2e}{h}$  is the unit of the quantum conductance and  $f(E - \mu_{L/R})$  is the Fermi–Dirac distribution function of the left/right electrode.

### 3. Results and discussion

# 3.1 Geometries of free-standing and graphene-supported Pd<sub>n</sub> clusters

The free-standing  $Pd_n$  clusters with 2–6 atoms are relaxed and the optimized structures are displayed in Fig. 1. As shown in Fig. 1, the average Pd–Pd bond length of a Pd<sub>n</sub> cluster increases with increasing its cluster size and gradually approaches to the bond length of 2.75 Å for the Pd bulk phase. The bond length



Fig. 1 Optimized geometries of free-standing  $Pd_n$  (n = 2-6) clusters. The average Pd-Pd bond lengths (in Å) are given for the clusters. Gray balls represent Pd atoms.

results are in good agreement with those values in literature.<sup>39</sup> For the  $Pd_n$  ( $n \ge 4$ ) clusters the three-dimensional (3D) structure is energetically more favorable than the flat one.

There are multiple ways to build a  $Pd_n$  cluster loaded on the graphene substrate. We have employed two different approaches to search for the most stable atomic configuration of a  $Pd_n$  cluster loaded on graphene. In the first approach, the optimized  $Pd_n$  clusters as shown in Fig. 1 are placed on the graphene surface in various orientations, and in the second approach the  $Pd_n$ /graphene structure is built by adding Pd atoms one by one to previously optimized  $Pd_n = \frac{1}{2}$ /graphene at

some possible positions. After relaxing all the atomic configurations, the most stable structures of Pd<sub>n</sub>/graphene are shown in Fig. 2 and the structural files are also provided as ESI.† Their structural and energetic results are presented in Table 1. In Table 1,  $d_{Pd-Pd}$  and  $d_{Pd-C(Gr)}$  are the average Pd–Pd bond length in the Pd<sub>n</sub> clusters and the shortest distance between the Pd atom and the C atom of graphene, respectively. A single Pd atom on the graphene surface prefers to occupy the bridge site as shown in Fig. 2(a). Similar to the single Pd atom case, for the Pd<sub>n</sub> (n = 2 & 3) clusters all Pd atoms prefer to occupy the bridge sites. For a larger Pd<sub>n</sub> (n = 4–6) cluster it tends to have a 3D



Fig. 2 The most stable configurations of  $Pd_n$  (n = 1-6) clusters supported on graphene are presented in (a–f). Brown and sliver balls represent the C and Pd atoms, respectively.

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**Table 1** Optimized geometric and energetic results of Pd<sub>n</sub> (n = 1-6) clusters supported on graphene. The average Pd–Pd bond length  $d_{Pd-Pd}$  in Pd<sub>n</sub> clusters and the average Pd–C bond length  $d_{Pd-C(Gr)}$  between Pd atom and adjacent C atom in graphene are also presented in the table

Clusters	$d_{\mathrm{Pd-Pd}}\left(\mathrm{\AA}\right)$	$d_{ m Pd-C(Gr)}( m \AA)$	$E_{\rm b}$ (eV)	$E_{\rm int}$ (eV)
Pd <sub>1</sub>	_	2.18	-1.10	-1.10
Pd <sub>2</sub>	2.71	2.20	-1.35	-1.44
Pd <sub>3</sub>	2.75	2.22	-1.52	-0.79
$Pd_4$	2.63	2.34	-1.85	-0.67
Pd <sub>5</sub>	2.66	2.41	-1.94	-0.66
Pd <sub>6</sub>	2.74	2.28	-2.07	-0.66

structure. The supported Pd5 and Pd6 clusters have very different atomic configurations from those free-standing ones, while the supported  $Pd_n$  (n = 2-4) clusters have the same shape as the free-standing ones although their average Pd-Pd bond lengths are enlarged. The  $Pd_n$  (n = 4-6) clusters have the shapes of tetrahedron, hexahedron and octahedron, respectively.<sup>39</sup> The Pd–Pd bond lengths in the Pd<sub>n</sub> clusters on the graphene surface vary in an irregular way and appear to oscillate as shown in Table 1, which is a general phenomenon well known for isolated metal clusters.<sup>40</sup> The distances d<sub>Pd-C(Gr)</sub> between the Pd atom and the C atom of graphene are also larger for the large  $Pd_n$  (n =4-6) clusters compared to those for the small  $Pd_n$  (n = 1-3)clusters, as shown in Table 1. This may be associated with the 2D–3D structure transformation of  $Pd_n$  clusters. Overall, our final stable  $Pd_n$  structures supported on the graphene substrate are very close to those obtained in ref. 39 and 40.

To evaluate the stability of a  $Pd_n$  cluster loaded on graphene, the binding energy  $E_b$  is calculated as follows

$$E_{\rm b} = [E_{\rm Pd_{\prime\prime}/Gr} - n \times E_{\rm Pd} - E_{\rm Gr}]/n \tag{4}$$

where  $E_{\text{Pd}_n/\text{Gr}}$  is same as it in eqn (1),  $E_{\text{Gr}}$  and  $E_{\text{Pd}}$  are the energies of the graphene supercell and free Pd atom, respectively. The interaction energy between the Pd<sub>n</sub> cluster and graphene is defined as

$$E_{\rm int} = E_{\rm Pd_n/Gr} - E_{\rm Pd_n} - E_{\rm Gr}$$
<sup>(5)</sup>

where  $E_{Pd_n}$  is the energy of the free-standing Pd<sub>n</sub> cluster, and the others on the right side of the equation are same as those in eqn (4). The interaction energy  $E_{int}$  is to evaluate the interaction between the Pd<sub>n</sub> cluster and the graphene substrate, while the binding energy  $E_b$  represents the stability of the system, *i.e.* the average formation energy per Pd atom.

We plot  $E_b$  and  $E_{int}$  as a function of the Pd<sub>n</sub> cluster size in Fig. 3. The binding energy is almost linearly dependent on the Pd cluster size. With increasing the cluster size, the cluster becomes more stable. The stability increase of the Pd<sub>n</sub> cluster supported on graphene is mainly attributed to the formation of Pd–Pd bonds. For example, for the Pd<sub>2</sub> cluster the binding energy (per Pd atom) is 0.25 eV lower than that of Pd<sub>1</sub> supported on graphene, which can be caused by the Pd–Pd bond formed in the cluster. For the interaction energy  $E_{int}$  in Fig. 3(b), it shows that the interaction between the Pd<sub>n</sub> cluster and graphene is stronger for the small Pd<sub>n</sub> (n = 1-3) cases relative to the large Pd<sub>n</sub> (n = 4-6) cases. The moderate interaction of about -0.66 eV for the large Pd<sub>n</sub> (n = 4-6) cases may explain their atomic structure changes and electron scattering effects upon HCHO adsorption, as discussed below. The interaction energy results are in agreement with the previous studies by Cabria *et al.*<sup>41</sup> and Xiao *et al.*<sup>42</sup>

#### 3.2 Formaldehyde adsorption on Pd(111) and Pd<sub>n</sub>/graphene

The formaldehyde adsorption on the Pd(111) surface is calculated, and the adsorption energy is defined as  $E_{ads}^{Pd(111)} = E_{HCHO-}$  $Pd(111) - E_{Pd(111)} - E_{HCHO}$ .  $E_{HCHO-Pd(111)}$  is the total energy of a formaldehyde molecule adsorbed on the Pd(111) slab, and  $E_{Pd(111)}$  and  $E_{HCHO}$  are the total energies of the Pd(111) slab and the isolated formaldehyde molecule, respectively. The slab model consisting of five Pd atomic layers and a vacuum thickness of 15 Å is constructed with the theoretical equilibrium lattice constant of 3.89 Å and a  $(3 \times 3)$  surface supercell. The adsorbed HCHO and Pd atoms in the top three layers are allowed to relax, while the atoms in the bottom two layers are fixed at the bulk lattice sites. After full structure relaxation, the most stable absorption geometry is obtained, in which the carbonyl group of HCHO gas molecule is parallel to the Pd(111) surface and both the C and O atoms bond to Pd atoms with the bond lengths of 2.17 Å and 2.14 Å, respectively. The calculated adsorption energy  $E_{ads}$  is -0.68 eV, which is at least 0.25 eV lower than those for different adsorption geometries, such as the one that only O atom is formed the bond with the Pd atom on the surface. The result is in good agreement with previous works.23,43,44

The adsorptions of HCHO molecule on the  $Pd_n$  (n = 1-6)clusters supported on graphene have been investigated. The initial HCHO adsorption geometries are built by considering various possible orientations of the HCHO molecule and adsorption sites. We fully relax the initial HCHO adsorption structures and obtain the most stable adsorption atomic configuration for each case. The optimized atomic configurations and the corresponding adsorption energies are presented in Fig. 4 and Table 2. In Fig. 4, we can see that the adsorbed HCHO molecule is bonded to the  $Pd_n$  clusters supported on graphene through both C and O atoms of HCHO molecule, which is similar to the HCHO adsorption on Pd(111).43 Moreover, the interaction of adsorbed HCHO molecule with the  $Pd_n$ cluster can dramatically affect the atomic configurations of the  $Pd_n$  clusters supported on graphene. For example, for the  $Pd_1$ case the Pd atom at the bridge site on graphene has been moved to the atop site upon HCHO adsorption, and the Pd atom in the top atomic layer of the Pd<sub>4</sub> cluster moves away from the vertex site to the bridge site between two Pd atoms. For the Pd<sub>5</sub> cluster the square base has been changed to the parallelogram shape, while for the Pd<sub>6</sub> cluster two Pd atoms in the top atomic layer move to the bridge sites from the hollow sites upon HCHO adsorption. However, we hope that the  $Pd_n$  clusters can resume their original atomic configurations in Fig. 2 after HCHO desorption and are applicable to HCHO detecting.



Fig. 3 The binding energy  $E_{\rm b}$  (a) and the interaction energy  $E_{\rm int}$  between the Pd<sub>n</sub> cluster and the graphene substrate (b) as a function of the cluster size.

**Table 2** Optimized atomic structures and adsorption energies of HCHO molecule absorbed on Pd<sub>n</sub>/graphene and Pd(111). The d<sub>C-O</sub> is the C-O bond length in the adsorbed HCHO molecule. The electron gains of the adsorbed HCHO molecule  $\Delta q_{HCHO}$  and graphene without HCHO adsorption  $\Delta q_{Grap}$  and with HCHO adsorption  $\Delta q_{Grap}$  (HCHO) are obtained by the Bader charge analysis, and the HCHO adsorption energies  $E_{ads}$  are calculated according to eqn (1)

Clusters	$d_{ ext{C-O}}\left( ext{\AA} ight)$	$\Delta q_{ m HCHO}$	$\Delta q_{ m Grap}$	$\Delta q_{ m Grap}$ (HCHO)	$E_{\rm ads}$ (eV)
-					
$Pd_1$	1.29	0.25	0.17	0.08	-1.42
$Pd_2$	1.32	0.36	0.24	0.19	-1.83
$Pd_3$	1.34	0.39	0.31	0.33	-1.98
$Pd_4$	1.33	0.36	0.26	0.29	-1.49
Pd <sub>5</sub>	1.32	0.34	0.30	0.32	-1.17
Pd <sub>6</sub>	1.31	0.32	0.30	0.31	-1.03
$Pd_{(111)}$	1.32	0.26	_	_	-0.68
. ,					

The HCHO adsorption energies on the  $Pd_n$  clusters supported on graphene are plotted in Fig. 5. The HCHO adsorption becomes stronger from Pd<sub>1</sub> to Pd<sub>3</sub> clusters, and then becomes weaker when further increasing the size of the  $Pd_n$  cluster. The strongest adsorption for the Pd<sub>3</sub> cluster supported on graphene has the adsorption energy of -1.98 eV. The adsorption on the 3D  $Pd_n$  clusters is weaker compared to those on the 2D  $Pd_n$ clusters, and it gradually approaches the adsorption energy of -0.68 eV on Pd(111) as the size of Pd<sub>n</sub> cluster increases. This large adsorption energy difference of 1.30 eV between the adsorptions on the Pd<sub>3</sub> cluster and Pd(111) makes it feasible to tune the HCHO adsorption strength on the  $Pd_n$  cluster decorated graphene by adjusting its size. The adsorption strength on the  $Pd_n$  (n = 5 and 6) cluster decorated graphene is in the proper adsorption energy range, implying their potential applications as HCHO sensors. The adsorption strength between the targeted gas and sensing materials is a key factor for sensor applications because it needs to balance the sensitivity and the recovery time. In a previous work,45 the authors investigated HCHO adsorption on Ti- and V-doped graphene and estimated the recovery times to be  $10^{-3}$  s and  $10^{4}$  s corresponding to the adsorption energies of -1.12 eV and -1.94 eV, respectively. Therefore, in our previous work<sup>24</sup> we suggested that the

adsorption energy should be in the range of -0.8 to -1.2 eV in order to have a good balance between the sensitivity and the recovery time. From Fig. 5, we also find that there is a certain correlation between the adsorption energy and C–O bond length and charge transfer of HCHO molecule. That is that for the Pd<sub>n</sub> cluster system which has a larger C–O bond length and charge transfer, the HCHO adsorption is also stronger, and *vice versa*. For the Pd<sub>3</sub> cluster, for instance, the HCHO adsorption is the strongest while it has the largest C–O bond length and charge transfer.

Compared to the C–O bond length (1.21 Å) of isolated HCHO molecule, the C-O bond of the adsorbed HCHO molecule is enlarged as shown in Table 2 and Fig. 5. The charge transfers of adsorbed HCHO are all positive and it means that the HCHO molecule acts as electron acceptor to gain the electrons from the Pd<sub>n</sub>/graphene substrate. We plot the C–O bond length and the charge transfer as a function of the size of  $Pd_n$  cluster in Fig. 5(b). As discussed in previous studies<sup>46</sup> the C-O bond length of the adsorbed HCHO molecule on the Pd<sub>n</sub>/graphene substrate is closely correlated with its electron gain from the substrate. From Fig. 5, it confirms that there is a similar trend for both the C-O bond length and the electron gain of adsorbed HCHO molecule. With increasing the size of  $Pd_n$  cluster, the C–O bond length and the electron gain of HCHO molecule first increase and then decrease. For the Pd<sub>3</sub> cluster case, the C-O bond expands 0.13 Å compared to that of HCHO molecule in the gas phase. The corresponding charge transfer of 0.39e is also the largest. A large elongation of the C-O bond implies the bond strength is weakened, which may be caused by the large charge transfer and strong interaction between the molecule and the Pd cluster.

The charge density difference (CDD) is defined as  $\Delta \rho = \rho_{\text{HCHO-Pd}_n/\text{Gr}} - \rho_{\text{HCHO}} - \rho_{\text{Pd}_n/\text{Gr}}$ , where  $\rho_{\text{HCHO-Pd}_n/\text{Gr}}$  is the charge density of the interacting system,  $\rho_{\text{HCHO}}$  and  $\rho_{\text{Pd}_n/\text{Gr}}$  are the charge densities of the isolated HCHO molecule and the Pd<sub>n</sub>/ graphene substrate, respectively. The 3D iso-surfaces of CDD and the planar-average charge-density difference  $\Delta \rho$  along the *z* axis are plotted in Fig. 6. The cyan and yellow areas of CDD denote the charge depletion and accumulation, respectively,

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Fig. 4 Optimized geometries of  $Pd_n$  (n = 1-6) cluster supported on graphene with HCHO adsorption are presented in (a-f). Brown, white, red and sliver balls represent the C, H, O and Pd atoms, respectively.



Fig. 5 (a) The HCHO adsorption energy on Pd<sub>n</sub> (n = 1-6) supported on graphene as a function of the cluster size. The dashed horizontal line in (a) represents the HCHO adsorption energy on Pd(111). (b) The C-O bond length  $d_{C-O}$  and charge transfer  $\Delta q$  of the adsorbed HCHO molecule on Pd<sub>n</sub>/graphene.

and the iso-surface level is set to 0.002 e Å<sup>-3</sup>. Positive values of the planar-average  $\Delta \rho$  indicate the electron gain at the *z*-axis transverse section. The adsorption of HCHO molecule onto Pd<sub>n</sub>/ graphene can greatly change the charge distribution. Obviously, there is an electron accumulation between the O or C atom of HCHO and the Pd atom that it is bonded to. From the planaraverage  $\Delta \rho$  for the Pd<sub>2</sub> and Pd<sub>3</sub> cases, it can clearly see that the largest electron gain is located the middle region of the C-Pd and O-Pd bonds. The CCD results in Fig. 6 verify that the HCHO molecule gains electrons from the substrate, which is consistent with the Bader charge analyses.

#### 3.3 Formaldehyde sensing properties of $Pd_n$ /graphene

The HCHO sensing properties of the  $Pd_n/graphene$  systems are carried out by the NEGF-DFT method.<sup>47,48</sup> Because of the anisotropy of graphene lattice, the nano-sensor devices are modeled for two different transport directions, *i.e.* the armchair- and zigzag-directions. The two-probe devices are consisted of the left/right electrode, the central scattering region that is made up with the  $Pd_n$  cluster supported on the graphene with/without HCHO adsorption. For the  $Pd_n$  clusters on graphene without and with HCHO gas adsorption the optimized atomic structures are presented in Fig. 2 and 4, and these



Fig. 6 The charge density differences for the HCHO molecule adsorbed on  $Pd_n/graphene$  (n = 2-5) together with the planar-average chargedensity  $\Delta \rho$  along the *z* axis are shown in (a–d). The iso-surface corresponds to the value of 0.002 e Å<sup>-3</sup>. The yellow and cyan regions represent charge accumulation and depletion, respectively. The dashed black, green, blue and red lines in the planar-average  $\Delta \rho$  mark the average positions of the graphene plane, Pd, C and O atoms of HCHO molecule in the *z* coordinates. The two dashed green lines in (c) and (d) refer to the two different Pd atomic layers.

stable structures do not need to optimize again since they are independent of the transport directions. The sensor models, using HCHO adsorption on Pd<sub>3</sub>/graphene as the example, are illustrated in Fig. 7. The atomic structures of the central scattering region are fully relaxed as discussed above.

The sensing performance can be evaluated in terms of the sensor response (S),<sup>49,50</sup> which is defined as the resistance difference after gas adsorption and before gas adsorption

divided by the resistance without gas adsorption. According to this definition, we have  $S = [(R_g - R_0)/R_0] \times 100\% = [(I_0 - I_g)/I_g] \times 100\%$ , in which the initial electronic resistance/current of the sensor without HCHO adsorption is  $R_0/I_0$  and the resistance/ current with HCHO adsorption is  $R_g/I_g$ . We have simulated the current–voltage (*I*–*V*) curves of all the Pd<sub>n</sub>/graphene cases with and without HCHO adsorption under the bias voltage from 0 to 1 V. The *I*–*V* and response results are presented in Fig. 8.



Fig. 7 Two-probe nano-sensor device models, using HCHO adsorption on  $Pd_3$ /graphene as the example. The semi-infinite left and right graphene electrodes (shaded zones) are in contact with the scattering region. Both the top and side views of the devices along the armchair (a) and zigzag (b) directions are shown. The C, Pd, O and H atoms are shown in brown, silver, red and white balls, respectively.

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Fig. 8 The current–voltage curves and responses of the armchair- and zigzag-orientated nano-sensor devices constructed from  $Pd_n/graphene$  (n = 1-6) under the bias voltage from 0 to 1.0 V are presented in (a–f).

The current increases as increasing the applied voltage for all the nano-sensors along both the armchair and zigzag transport directions. For the HCHO molecule adsorption on  $Pd_n/graphene$  (n = 1-3), it is found that the currents passing through the nano-devices are almost same before and after HCHO

adsorption in the whole applied voltage range. So the responses for these nano-sensors are insensitive regarding the HCHO gas adsorption. As shown in Fig. 8(d-f) for the large Pd<sub>n</sub> ( $n \ge 4$ ) cluster cases, the electronic current of the sensor with HCHO adsorption starts to differ from that without HCHO adsorption.

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The current is increased upon the HCHO adsorption relative to the case without HCHO adsorption. The corresponding responses are also substantially larger than those for the  $Pd_n$  (n = 1-3) cluster systems. We can see that the sensor response depends on the applied voltage and generally decreases as the applied voltage increases. The dependence of response on the applied voltage will be discussed below. A large response at the small bias voltage implies that the device has a short response time and a high sensitivity. Therefore, these devices with the highest absolute response over 20% at small bias voltages should be sensitive enough for the HCHO gas detecting.<sup>51-53</sup> Especially, for the armchair-orientated  $Pd_5$ /graphene nanosensor it has the highest absolute response over 60% at 0.2 V, indicating its suitability of HCHO sensing.

The transmission spectra in the equilibrium situation with zero bias for the  $Pd_n$ /graphene substrate adsorbed with/without HCHO molecule are plotted in Fig. 9. The transmission spectra exhibit a dependence on the size and geometry of the  $Pd_n$  cluster. For the  $Pd_n$  (n = 1–3) cases, the transmission spectra near zero bias are relatively smoother compared to those for the larger  $Pd_n$  (n = 4–6) clusters and almost same before and after HCHO adsorption. This explains why the nano devices consisted of the  $Pd_n$  (n = 1–3) clusters are insensitive with respect to the HCHO adsorption for these cases. For the larger  $Pd_n$  (n = 4–6) cases we see that the transmission spectra are quite different on the left side of zero bias before and after HCHO adsorption. The transmission spectra vary irregularly without HCHO adsorption, which is mainly attributed to the electron scattering

effect of the complex local deformation of graphene loaded with a larger Pd<sub>n</sub> cluster. Upon the HCHO adsorption, the interaction between the  $Pd_n$  cluster and graphene is weakened caused by the interaction between the HCHO and the  $Pd_n$  cluster supported on graphene, and consequently the electron scattering effect of graphene local deformation becomes relatively weaker compared to the case without HCHO adsorption. From Table 2 we can see that for the  $Pd_n$  (n = 3-6) cases the graphene electron gains before and after HCHO adsorption have a very small change. The electron gain of graphene for the Pd3 case is similar to those for other  $Pd_n$  (n = 4-6) cases, but the current increases are quite different for all the four cases. This indicates that the current change after HCHO adsorption can be mainly attributed to the graphene scattering effect, rather than the number change of charge carriers. The transmission coefficients for the HCHO adsorption cases are generally larger than those without HCHO adsorption, which is consistent with the current-voltage results in Fig. 8.

In order to understand the larger response under low bias voltages, the transmission spectra of the nano-sensor constructed from  $Pd_5$ /graphene along the armchair-direction are presented in Fig. 10. We see that within the voltage window (indicated by the dashed lines) the transmission spectra differ from each other before and after HCHO adsorption. The transmission spectrum change before and after HCHO adsorption, and the integration of the transmission spectrum within the voltage window determines the current. According to the definition of



Fig. 9 Transmission spectra for the  $Pd_n$ /graphene (n = 1-6) sensors without and with HCHO adsorption at zero bias are shown in (a–f). The dashed lines, which are set to be zero on the energy scale, indicate the Fermi level.



Fig. 10 The transmission spectra of HCHO nano-sensor constructed from  $Pd_5$ /graphene along the armchair-direction under (a-c) 0, 0.4 and 0.8 V bias. The dashed lines represent the bias windows.

response, it is the ratio of the current difference before and after gas adsorption to the current without gas adsorption. Since the current is low under low bias voltage the response of nanosensor under low bias voltages can be larger than those under high bias voltages.

### 4. Conclusions

The HCHO sensing properties of small  $Pd_n$  (n = 1-6) cluster decorated graphene have been systematically investigated by performing first-principles calculations in the framework of density-functional theory. The optimized geometries of Pd<sub>n</sub> cluster decorated graphene with and without HCHO adsorption, adsorption energies and charge transfers are obtained and analyzed for various Pd<sub>n</sub> cluster cases. The adsorption strength of HCHO gas molecule on the  $Pd_n$  cluster decorated graphene is dependent on the cluster size and can be tuned in a wide range from -0.68 eV on Pd(111) to -1.98 eV on the Pd<sub>3</sub> cluster decorated graphene. The calculated results show that the  $Pd_n$  (*n* = 5 and 6) cluster decorated graphene systems have a proper adsorption energy for the HCHO sensing. The adsorbed HCHO gas molecule gains electrons from the  $Pd_n$  cluster decorated graphene, acting as an acceptor. The HCHO adsorption strength is found to be correlated with the electron gain and C-O bond length. The two-probe sensor devices consisting of the central scattering region and pristine graphene electrodes with the armchair and zigzag transport directions are built, and the current-voltage curves are calculated by the nonequilibrium Green's function method. The results indicate that the devices constructed from the large  $Pd_n$  ( $n \ge 4$ ) cluster decorated graphene are sensitive enough for HCHO detecting with the highest absolute response value over 20% at small bias voltages. Combined with the adsorption energy results we consider that the devices constructed from the  $Pd_n$  (n = 5 and 6) cluster decorated graphene should be suitable for HCHO gas

sensing. The present work provides a useful guidance for designing the HCHO gas sensors based on small metal cluster decorated graphene.

### Conflicts of interest

The authors declare no academic misconducts or competing financial interest.

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### References

- 1 K. Kawamura, K. Kerman, M. Fujihara, N. Nagatani, T. Hashiba and E. Tamiya, *Sens. Actuators, B*, 2005, **105**, 495–501.
- 2 T. Salthammer, S. Mentese and R. Marutzky, *Chem. Rev.*, 2010, **110**, 2536–2572.
- 3 R. D. Timothy, N. R. Carey, P. Susan, C. D. Glass, B. Geza, R. Alison and F. Lin, *Ann. Occup. Hyg.*, 2015, **60**, 132–138.
- 4 S. Suresh and T. J. Bandosz, Carbon, 2018, 137, 207-221.
- 5 J. K. McLaughlin, Int. Arch. Occup. Environ. Health, 1994, 66, 295–301.
- 6 A. Duong, C. Steinmaus, C. M. McHale, C. P. Vaughan and L. Zhang, *Mutat. Res., Rev. Mutat. Res.*, 2011, **728**, 118–138.
- 7 Q. Huang, D. Zeng, H. Li and C. Xie, *Nanoscale*, 2012, 4, 5651–5658.
- 8 P. R. Chung, C. T. Tzeng, M. T. Ke and C. Y. Lee, *Sensors*, 2013, **13**, 4468–4484.
- 9 D. Chen and Y. J. Yuan, IEEE Sens. J., 2015, 15, 6749-6760.
- 10 I. Cabria, M. J. López, S. Fraile and J. A. Alonso, *J. Phys. Chem. C*, 2012, **116**, 21179–21189.
- 11 R. Hussain, M. Saeed, M. Y. Mehboob, S. U. Khan, M. Usman Khan, M. Adnan, M. Ahmed, J. Iqbal and K. Ayub, *RSC Adv.*, 2020, **10**, 20595–20607.
- 12 A. Safavi, N. Maleki, F. Farjami and E. Farjami, *J. Electroanal. Chem.*, 2009, **626**, 75–79.
- 13 F. Niu and Q. Yi, Rare Met., 2011, 30, 102-105.
- 14 H. Yin, H. Tang, D. Wang, Y. Gao and Z. Tang, *ACS Nano.*, 2012, **6**, 8288–8297.
- 15 D. Sen, R. Thapa and K. K. Chattopadhyay, *Int. J. Hydrog. Energy*, 2013, **38**, 3041–3049.
- 16 A. Gutés, B. Hsia, A. Sussman, W. Mickelson, A. Zettl, C. Carraro and R. Maboudian, *Nanoscale*, 2012, 4, 438–440.
- 17 W. Dong, Y. Ren, Z. Bai, Y. Yang and Q. Chen, *Bioelectrochem*, 2019, **128**, 274–282.
- 18 R. Kumar, D. Varandani, B. R. Mehta, V. N. Singh, Z. Wen, X. Feng and K. Mullen, *Nanotechnology*, 2011, 22, 275719.
- 19 Y. Y. Fan, H. L. Tu, Y. Pang, F. Wei, H. B. Zhao, Y. Yang and T. L. Ren, *Rare Met.*, 2020, **39**, 651–658.
- 20 Q. Yi, F. Niu and W. Yu, *Thin Solid Films*, 2011, **519**, 3155–3161.

- 21 K. Czelej, K. Cwieka, J. C. Colmenares and K. J. Kurzydlowski, *Langmuir*, 2016, **32**, 7493–7502.
- 22 M. Yoosefian, H. Raissi and A. Mola, *Sens. Actuators, B*, 2015, 212, 55–62.
- 23 J. Deng, W. Song, M. Jing, T. Yu, Z. Zhao, C. Xu and J. Liu, *Catal. Today*, 2020, **339**, 210–219.
- 24 L. Yang, W. Xiao, J. Wang, X. Li and L. Wang, *J. Mater. Sci.*, 2021, **56**, 12256–12269.
- 25 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, 6, 15–50.
- 26 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169.
- 27 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, 77, 3865.
- 28 G. Kresse and D. Joubert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758–1775.
- 29 S. Grimme, J. Comput. Chem., 2006, 27, 1787-1799.
- 30 J. Zhou, Q. Wang, Q. Sun, X. Chen, Y. Kawazoe and P. Jena, *Nano Lett.*, 2009, **9**, 3867–3870.
- 31 G. Henkelman, A. Arnaldsson and H. Jónsson, *Comput. Mater. Sci.*, 2006, 36, 354–360.
- 32 W. Tang, E. Sanville and G. Henkelman, *J. Phys.: Condens. Matter*, 2009, **21**, 084204.
- 33 J. Taylor, H. Guo and J. Wang, Phys. Rev. B: Condens. Matter Mater. Phys., 2001, 63, 245407.
- 34 D. Waldron, P. Haney, B. Larade, A. MacDonald and H. Guo, *Phys. Rev. Lett.*, 2006, **96**, 166804.
- 35 M. Brandbyge, N. Kobayashi and M. Tsukada, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **60**, 17064–17070.
- 36 M. Brandbyge, J. L. Mozos, P. Ordejón, J. Taylor and K. Stokbro, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, 65, 165401.
- 37 S. Datta, *Quantum transport: atom to transistor*, Cambridge University Press, 2005.

- 38 M. Büttiker, Y. Imry, R. Landauer and S. Pinhas, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1985, **31**, 6207–6215.
- 39 J. Rogan, G. García, J. A. Valdivia, W. Orellana, A. H. Romero, R. Ramírez and M. Kiwi, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, 72, 115421.
- 40 J. A. Alonso, *Structure and properties of atomic nanoclusters*, Imperial College Press, London, 2005.
- 41 I. Cabria, M. J. López and J. A. Alonso, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 035403.
- 42 W. Xiao, R. Zeng, L. Cheng, J. W. Wang, L. J. Jiang and L. G. Wang, *RSC Adv.*, 2015, 5, 61861–61867.
- 43 E. Jeroro and J. M. Vohs, J. Am. Chem. Soc., 2008, 130, 10199-10207.
- 44 X. K. Gu and W. X. Li, J. Phys. Chem. C, 2010, 114, 21539– 21547.
- 45 X. Chen, L. Xu, L. L. Liu, L. S. Zhao, C. P. Chen, Y. Zhang and X. C. Wang, *Appl. Surf. Sci.*, 2017, **396**, 1020–1025.
- 46 M. Zhou, Y. H. Lu, Y. Q. Cai, C. Zhang and Y. P. Feng, *Nanotechnology*, 2011, 22, 385502.
- 47 D. B. Farmer, R. Golizadeh Mojarad, V. Perebeinos, Y. M. Lin, G. S. Tulevski, J. C. Tsang and P. Avouris, *Nano Lett.*, 2009, 9, 388–392.
- 48 L. Kou, T. Frauenheim and C. Chen, J. Phys. Chem. Lett., 2014, 5, 2675–2681.
- 49 K. Lee, R. Gatensby, N. McEvoy, T. Hallam and G. S. Duesberg, *Adv. Mater.*, 2013, **25**, 6699–6702.
- 50 S. Yang, C. Jiang and S. h. Wei, *Appl. Phys. Rev.*, 2017, 4, 021304.
- 51 A. J. Yang, D. W. Wang, X. H. Wang, J. F. Chu, P. L. Lv, Y. Liu and M. Z. Rong, *IEEE Electron Device Lett.*, 2017, 38, 963–966.
- 52 S. J. Young and Z. D. Lin, *ECS J. Solid State Sci. Technol.*, 2017,
   6, M130–M132.
- 53 F. Niu, L. M. Tao, Y. C. Deng, Q. H. Wang and W. G. Song, *New J. Chem.*, 2014, 38, 2269–2272.