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Investigation on the upconversion luminescence and ratiometric thermal sensing of SrWO₄:Yb³⁺/ RE³⁺ (RE = Ho/Er) phosphors

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 $SrWO_4$ phosphors doped with $Ho^{3+}(Er^{3+})/Yb^{3+}$ are successfully prepared by a high temperature solid-state reaction method. The upconversion (UC) luminescence properties of all the samples have been investigated under 980 nm excitation. Strong green emissions are obtained in the SrWO₄:Yb³⁺/Ho³⁺ and SrWO₄:Yb³⁺/ Er³⁺ samples with the naked eyes. In a temperature range going from 303 K to 573 K, the UC emission spectra of the phosphors have been measured. Then the temperature sensing properties also have been discussed via fluorescence intensity ratio (FIR) technology. For the SrWO₄:Yb³⁺/Ho³⁺ phosphor, the FIR technologies based on thermal coupling levels (TCLs)(5F4,5F5) and non-thermal coupling levels (non-TCLs)(${}^{5}S_{2}$, ${}^{5}F_{4}$ / ${}^{5}F_{5}$) are used for investigating the sensitivity. The results show that the maximum absolute sensitivity reaches 0.0158 K^{-1} with non-TCLs. As for Yb³⁺/Er³⁺ codoped SrWO₄ phosphor, the maximum absolute sensitivity reaches 0.013 K⁻¹ with TCLs (${}^{2}H_{11/2}$, ${}^{4}S_{5/2}$) at a temperature of 513 K. These significant results demonstrate that the SrWO₄:Ho $^{3+}$ (Er $^{3+}$)/Yb $^{3+}$ phosphors are robust for optical temperature sensors.

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Introduction

As the basic parameter of thermodynamics, temperature measurement plays a vital role in scientific research, industrial production and medicine.1 In recent years, much attention has been paid to non-contact temperature sensors based on rare earth (RE) ion activated luminescent materials.2 Some UC luminescent materials with doping of RE3+ ions have been studied for temperature sensors.3-7 The FIR technology based on the measurement of the temperature-dependent FIR from two excited state energy levels of activators represents an accurate temperature measuring method, because the FIR from the thermally coupled levels (TCLs) of RE ions is independent of the excitation intensity fluctuations, external disturbance and spectrum losses. Thus, reasonable measurement accuracy and sensitivity could be obtained with this method. The variation of FIR is generally caused by a thermally induced population redistribution among the TCLs. And some RE ions have already been explored for designing FIR thermometry, such as Er3+, Tm³⁺, Ho³⁺ et al.⁸⁻¹³ The sensitivity is an important role of a temperature sensing device. The absolute sensitivity is defined as the rate of change of FIR with change in temperature relative to FIR. Basically, a larger energy gap between the TCLs benefits the enhancement of sensitivity. However, the largest energy mismatches between those TCLs cannot exceed 2000 cm⁻¹, so it is difficult for the further improvement of sensitivity. There is an urgent need to design a new temperature

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measurement method to obtain high detecting sensitivity. Recently, a new kind of strategy has been proposed to solve the above drawback, which is based on FIR between the non-TCLs of the activators. The FIR derived from non-TCLs is also related to temperature. Compared with the FIR technique based on the TCLs, the FIR technique based on the non-TCLs is no longer limited by the energy gap. The Er³⁺ and Ho³⁺ ions also have non-TCLs pairs in addition to TCLs. So further investigation on temperature sensing properties of Er³⁺ and Ho³⁺ ions based on the FIR technology (TCLs and non-TCLs) is necessary.

Besides activators, temperature sensing properties of UC luminescent materials are also decided by host material. Among various hosts, tungstates have obtained much attention because of their brilliant physical and chemical stability, low phonon threshold energy, and high density.14 The Ln3+ ions also have high doped concentration in tungstates host. So the activators can emit intense and stable fluoresce even at high temperature.15 SrWO4 belongs to a body-centered tetragonal system with scheelite crystal structure where WO₄²⁻ molecular ions are loosely bound to Sr²⁺ cations. It has been reported that SrWO₄ is an ideal host for optical temperature sensing materials. Recently, the temperature sensing properties of Ln3+ doped tungstates also have been studied. Pandey et al. report that the Er³⁺-Yb³⁺ co-doped SrWO₄ phosphor is a good optical temperature sensing material, and its maximum absolute sensitivity reaches 0.01498 K⁻¹.16 The high sensitivity of Tm³⁺/ Yb3+ co-doped SrWO4 phosphor for optical thermometry is synthesized by Song et al., and when the temperature is 323 K, the absolute sensitivity reaches a maximum of 0.00617 K⁻¹.17 For Sm3+ doped SrWO4 phosphor and Nd3+/Yb3+ co-doped

SrWO₄ phosphor, they also have good temperature sensing performance. ^{18,19} Note that, the SrWO₄ material is more suitable to be a UC fluorescence host for designing optical temperature sensor. However, as far as we know, the SrWO₄:Yb³⁺/Ho³⁺ phosphor for optical thermometry has not been investigated. There are even fewer reports on its optical thermometry using FIR technique based on the non-TCLs.

In this paper, the SrWO₄:Yb³⁺/RE³⁺(RE = Er, Ho) phosphors are synthesized by a high temperature solid state reaction. The UC luminescence performances and temperature sensing properties are investigated under 980 nm excitation. The experimental results illustrate that the temperature sensitivity of our phosphors are better than the most reported $\mathrm{Ho^{3+}(Er^{3+})}$ doped materials. The SrWO₄ doping $\mathrm{Ho^{3+}(Er^{3+})}$ ions materials are optical temperature sensing material with good application value.

2. Experimental

2.1 Synthesis of the phosphor

The samples are designed according to the molar composition of $Sr_{1-x-0.01}Yb_xHo_{0.01}WO_4$ (x=0, 1%, 2%, 3%, 4%, 6%, 10%) and $Sr_{1-y-0.01}Yb_yEr_{0.01}WO_4$ (y=0, 1%, 2%, 3%, 4%), and synthesized by solid-state reaction method. High-purity $SrCO_3$ (99%), WO_3 (99.8%), Yb_2O_3 (99.99%), Ho_2O_3 (99.99%) and Er_2O_3 (99.99%) are purchased from Aladdin Chemical Reagent Co. Ltd (China). The raw materials are weighed by molar ratio and mixed together. The mixtures thoroughly are ground in a mortar of agate for 60 minutes. The ground mixture is transferred to a crucible ceramics and then heated it in a muffle furnace. First, the temperature rises to 900 °C by 10 °C min⁻¹. The samples are kept at 900 °C for 4 h. Next, temperature is increased by 5 °C min⁻¹ to 1100 °C and kept for 6 h. Finally the $SrWO_4$: Ho^{3+}/Yb^{3+} and $SrWO_4$: Er^{3+}/Yb^{3+} phosphors are obtained.

2.2 Measurement and characterization

The structural formation of the phosphors has been measured using an X-ray diffraction (Rigaku D/Max-2500) in the range of 15° to 80° (2 θ), and its radiation source is Cu K α ray of λ = 0.15406 nm. For morphology and size of the phosphors, the

field emission scanning electron microscope (FE-SEM, JEOL JEM-6700F) is used to perform. The UC emission spectra are recorded using the Zolix Omni- λ 500 spectrometer under a 980 nm laser (MDL-III-980-2W, China) excitation. The samples are heated using an Orient KOJI TAP-02 high temperature thermometer, among a temperature range of 303 to 573 K, with a temperature control accuracy of 0.1 °C.

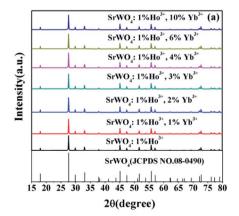
3. Result and discussion

3.1 Structural characterization

Fig. 1(a) shows the XRD patterns of the SrWO₄:Ho³⁺/Yb³⁺ phosphors. It can be seen that the diffraction peaks of the samples are very consistent with the standard card of SrWO₄ (JCPDS no. 08-0490). The sharp and strong diffraction peaks show that the synthesized samples are crystallized well. It also means that Sr2+ in the SrWO4 lattice is substituted by Ho3+ and Yb³⁺ ions. The XRD patterns of the SrWO₄:Er³⁺/Yb³⁺ samples are shown in Fig. 1(b). The main diffraction peaks of samples are also consistent with the standard card of SrWO₄ (JCPDS no. 08-0490). But an extra weak peak at 29.3° appears in SrWO₄:1% Er^{3+} , x%Yb³⁺(x = 2, 3, 4) samples, which may be induced by the minor unreacted component of Yb₂O₃. Once the concentration of the Yb³⁺ ions is over 2%, a part of the Yb³⁺ ions will not be added in the lattice. In addition, the peak at 30° moves to a high angle direction with the concentration of Yb³⁺ ions increasing (see the illustration in Fig. 1(b)), which is attributed to the shrinkage of the crystal lattice caused by the doped Er³⁺ and Yb³⁺ ions substituting the Sr²⁺ sites. The ionic radius of Er³⁺, Yb³⁺ and Sr²⁺(Er³⁺: 0.89 Å, Yb³⁺: 0.86 Å and Sr²⁺: 1.12 Å)are different, so the crystal lattice will produce deformation. The SEM images of SrWO₄:Ho³⁺(Er³⁺)/Yb³⁺ phosphor are shown in Fig. 2. It indicates that the size of the prepared phosphor is almost uniform.

3.2 Upconversion emission properties

Fig. 3(a) shows the UC emission spectrum of $SrWO_4:1\%Ho^{3+}/2\%Yb^{3+}$ phosphor from 500 to 800 nm under 980 nm excitation. Three characteristic emission bands are presented that the strong green emission (\sim 540 nm) belongs to the radiative



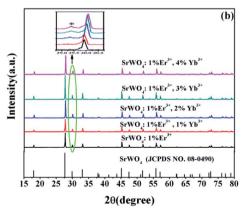


Fig. 1 XRD patterns of SrWO₄: (a) $1\% Ho^{3+}/x\% Yb^{3+}(x=0, 1, 2, 3, 4, 6, 10)$ phosphors. (b) $1\% Er^{3+}/x\% Yb^{3+}(x=0, 1, 2, 3, 4)$ phosphors.

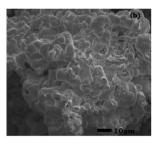


Fig. 2 SEM images of SrWO₄: (a) 1%Ho³⁺/2%Yb³⁺ phosphor. (b) 1%Er³⁺/1%Yb³⁺ phosphor.

transition from the ${}^5S_2/{}^5F_4$ to 5I_8 , the red emission (\sim 663 nm) belongs to the radiative transition from 5F_5 to 5I_8 and the weak near infrared emission (\sim 756 nm) corresponds to ${}^5F_4 \rightarrow {}^5I_7$ energy level transition of Ho3+ ions. Fig. 3(b) shows the UC emission spectra of SrWO₄:1%Ho³⁺/x%Yb³⁺ (x = 0, 1, 2, 3, 4, 6,10). It is seen that, the intensities of all the UC emission peaks reach maximum values at x = 2. But they decrease if the Yb³⁺ concentration is further increased up to 2%. The Yb3+ ion is a high-efficiency sensitizer for Ho3+ ion because it has lager absorption cross-sectional area at the near infrared. With increasing the Yb3+ ions concentration to 2%, the energy transfer process from Yb3+ to Ho3+ can be enhanced, resulting in higher UC emission intensity. However, with the further enhancement of the Yb3+ concentrations, the distance between the doping ions decreases with the increase of Yb3+ concentration. It induces more intense interaction among adjacent Yb3+ ions, which cause concentration quenching.20

In order to explicate the possible UC emission process of the SrWO₄:Ho³⁺/Yb³⁺ sample, the dependence relation of the UC

emission intensity via pump power is measured. The relationship between UC emission intensity and pumping power can be described by the following eqn (1):²¹

$$I \propto P^n$$
 (1)

where I is the UC emission intensity, P is the pumping power and n is the photon numbers which correspond to populate the upper emitting levels. Fig. 3(c) shows the log–log plot of intensity *versus* pump power and the slopes of the fitted lines of SrWO₄:1%Ho³⁺/2%Yb³⁺ sample. It can be seen that n values are 1.21, 1.24 and 2.27 for green (\sim 540 nm), red (\sim 663 nm), and near-infrared (\sim 756 nm) emissions. It indicates that these UC emission processes are related to the two-photon processes. The energy level diagrams of Ho³⁺/Yb³⁺ ions and possible UC emission processes are shown in Fig. 3(d).

Under 980 nm excitation, the Yb3+ ion are excited from the ground state ²F_{7/2} to the excited state ²F_{5/2} by ground state absorption (GSA) process. Ho³⁺ ions are excited to the excited state levels mainly through the energy transfer (ET) process from Yb³⁺ to Ho³⁺. Through ET1 (${}^{2}F_{5/2}$ (Yb³⁺) + ${}^{5}I_{8}$ (Ho³⁺) $\rightarrow {}^{2}F_{7/2}$ $(Yb^{3+}) + {}^{5}I_{6}$ (Ho³⁺)), the Ho³⁺ ions are excited from the ${}^{5}I_{8}$ to the ⁵I₆ state.²² Some Ho³⁺ ions at ⁵I₆ state can nonradiatively (NR) relax to ⁵I₇ state, and then populating the ⁵F₅ state by excited state absorption (ESA) from ⁵I₇ to ⁵F₅. Ho³⁺ ions at ⁵I₆ state can continue to be excited into the 5S2, 5F4 coupling state by ET2 $({}^{2}F_{5/2} (Yb^{3+}) + {}^{5}I_{6} (Ho^{3+}) \rightarrow {}^{2}F_{7/2} (Yb^{3+}) + {}^{5}S_{2}, {}^{5}F_{4} (Ho^{3+})).$ Subsequently, the Ho³⁺ ions at the ⁵S₂, ⁵F₄ coupling state will relax to the ⁵F₅ state by NR relaxation, and then a part of the Ho³⁺ ions at the ⁵F₅ state are transferred to the ⁵I₈ state, concurrently red light of 663 nm is emitted. The 5S2, 5F4 coupling state relax to the 5I8 and 5I7 states, resulting in high

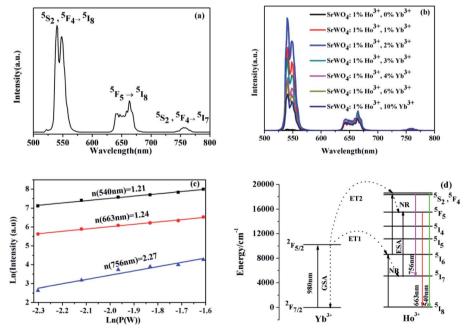


Fig. 3 (a). The emission spectrum of $SrWO_4:1\%Ho^{3+}/2\%Yb^{3+}$ phosphor from 500 to 800 nm under exciting at 980 nm. (b). The UC emission spectra of $SrWO_4:1\%Ho^{3+}/x\%Yb^{3+}$ (x=0,1,2,3,4,6,10). (c). Log-log diagram of excitation power density and UC emission intensity. (d). The energy level diagrams of Ho^{3+}/Yb^{3+} ions and possible UC emission processes.

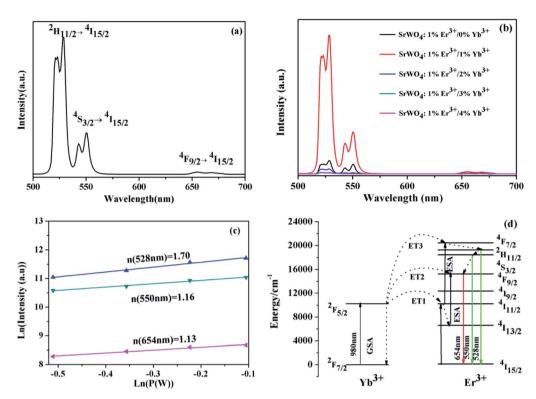


Fig. 4 (a). The UC emission spectrum of SrWO $_4$:1%Er 3 +/1%Yb 3 + phosphor. (b). UC emission spectra of SrWO $_4$:1%Er 3 +/x%Yb 3 + (x = 0, 1, 2, 3, 4). (c). Log-log relationship between pump power and UC emission intensity of SrWO $_4$:1%Er 3 +/1%Yb 3 + phosphor. (d). The energy level diagrams of Er 3 +/Yb 3 + ions and possible UC emission processes.

intensity green light (\sim 540 nm) and low intensity red light (\sim 756 nm), respectively.

The UC emission spectrum of SrWO₄:1%Er³⁺/1%Yb³⁺ phosphor is shown in Fig. 4(a). Two UC emission bands are exhibited, the strong green and the weak red emission, which can be attributed to the radiative transitions from ${}^{2}H_{11/2}/{}^{4}S_{3/2}$ to ${}^{4}I_{15/2}$ and the ${}^4I_{9/2} \rightarrow {}^4I_{15/2}$ of the Er³⁺ ions, respectively. Fig. 4(b) shows the UC emission spectra of SrWO₄:1%Er³⁺/x%Yb³⁺ (x = 0, 1, 2, 3, 4). It can be found that the intensities of two emission bands first increase and then decrease with increasing Yb3+ concentrations, and reach a maximum at x = 1. Once the concentration of Yb3+ is over 1%, the UC emission intensities rapidly decline. This is mainly attributed to that the energy transfer efficiency from Yb3+ to Er3+ ion decrease because the limit of the Yb3+ that can be stabilized into the matrix. As demonstrated by the extra peaks in the XRD patterns, when the concentration of the Yb³⁺ ions is over 2%, a part of the Yb³⁺ ions will not be added in the lattice.

The dependence relationship of the UC emission intensities of SrWO₄:Er³+/Yb³+ on pump power are shown in Fig. 4(c). For the emission $\sim\!528$ nm, 550 nm and 654 nm, the slopes of the fitting experiment data are 1.70, 1.16 and 1.13, which is near 2. It indicates that all emission processes are related to the two-photon processes. Fig. 4(d) shows the energy level diagrams and possible UC emission processes of Er³+ and Yb³+ ion. Under 980 nm laser excitation, Yb³+ ions are excited from the ground state $^2F_{7/2}$ to the excited state $^2F_{5/2}$ by GSA process. The Er³+ ions are excited to the $^4I_{11/2}$ state from the ground state through the

ET1 (${}^2F_{5/2}$ (Yb ${}^{3+}$) + ${}^4I_{15/2}$ (Er ${}^{3+}$) $\rightarrow {}^2F_{7/2}$ (Yb ${}^{3+}$) + ${}^4I_{11/2}$ (Er ${}^{3+}$)). After that, the Er ${}^{3+}$ ions at ${}^4F_{7/2}$ state can be populated from the ${}^4I_{11/2}$ state by the ET3 (${}^2F_{5/2}$ (Yb ${}^{3+}$) + ${}^4I_{11/2}$ (Er ${}^{3+}$) $\rightarrow {}^2F_{7/2}$ (Yb ${}^{3+}$) + ${}^4F_{7/2}$ (Er ${}^{3+}$)). The ${}^2H_{11/2}$ and ${}^4S_{3/2}$ states are populated from the ${}^4F_{7/2}$ state owing to the NR relaxation, and then the Er ${}^{3+}$ ions at ${}^2H_{11/2}$ and ${}^4S_{3/2}$ states relax to the ${}^4I_{15/2}$ state, leading to the green UC emissions (528 nm and 550 nm). The Er ${}^{3+}$ ions at ${}^4I_{11/2}$ state can relax to the ${}^4I_{13/2}$ state through the multiphonon relaxation (MPR) process, and then the Er ${}^{3+}$ ions at ${}^4I_{13/2}$ state are excited to the ${}^4F_{9/2}$ by ET2 (${}^2F_{5/2}$ (Yb ${}^{3+}$) + ${}^4I_{13/2}$ (Er ${}^{3+}$) $\rightarrow {}^2F_{7/2}$ (Yb ${}^{3+}$) + ${}^4F_{9/2}$ (Er ${}^{3+}$)). Meanwhile, the ${}^4F_{9/2}$ state can be also populated from the ${}^4S_{3/2}$ state through the NR relaxation. The Er ${}^{3+}$ at the ${}^4F_{9/2}$ state will transit to the ${}^4I_{15/2}$ state and thus emitted the red light (\sim 654 nm).

3.3 Optical temperature-sensing properties

For investigating the optical temperature sensing characteristics of SrWO₄:Ho³⁺/Yb³⁺ phosphor, the emission spectra of the SrWO₄:Ho³⁺/Yb³⁺ sample are measured at different temperature. As shown in Fig. 5(a), it can be seen that the UC emission intensities of all emission band decrease with the temperature increasing. The possible reason is thermal quenching effect.²³ But the decreasing rate of the different emission band is different. Fig. 5(b) shows the UC emission intensity variation of 540 nm (5 S₂, 5 F₄ \rightarrow 5 I₈) and 756 nm (5 S₂, 5 F₄ \rightarrow 5 I₇) with temperature increasing. It is clearly observed that the UC emission intensity at 756 nm drops much more slowly than that at 540 nm with the temperature increasing. By considering the



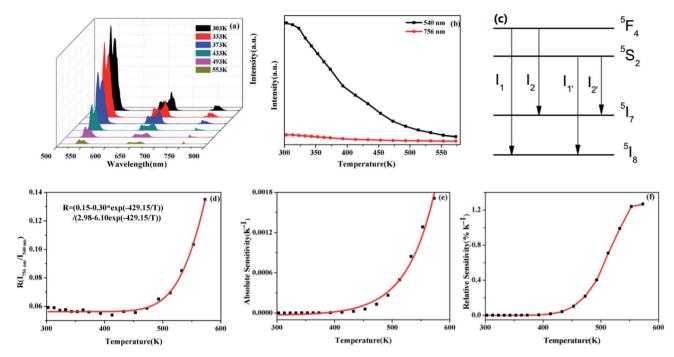


Fig. 5 (a). Temperature-dependent on UC emission spectra of SrWO₄:1%Ho³⁺/2%Yb³⁺ phosphor at 303–573 K. (b). The emission intensity of SrWO₄:1%Ho³⁺/2%Yb³⁺ phosphor center at 540 nm and 756 nm change in the range 303-573 K. (c) Four-level simplified diagram of transitions from adjacent excited levels to two lower levels. (d) The FIR $(I_{756 \text{ nm}}/I_{540 \text{ nm}})$, (e) the absolute sensitivity based on the $I_{756 \text{ nm}}/I_{540 \text{ nm}}$ (f) the relative sensitivity based on the I_{756 nm}/I_{540 nm}.

thermalization between 5S2 and 5F4 levels, the FIR of the UC emission from $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{8}$ and $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{7}$ transitions can be analyzed by using a four-level system, which was introduced by González-Pérez et al.²⁴ as shown in Fig. 5(c). The ⁵S₂, ⁵F₄, ⁵I₇ and ⁵I₈ levels form a four-level system. The emission intensity is proportional to the population of each energy level and the population of the excited state relates to the temperature. So the FIR can be expressed by the following eqn (2).25

$$FIR_{1} = \frac{I_{756 \text{ nm}}}{I_{540 \text{ nm}}} = \frac{C_{1} + C_{2} \exp(-\Delta E/kT)}{C_{3} + C_{4} \exp(-\Delta E/kT)}$$
(2)

where C_1 , C_2 , C_3 and C_4 are constants about spontaneous emission rate, energy level degeneracy and emission energy. ΔE is the energy gap of the 5S_2 and 5F_4 , k is the Boltzmann constant, and T is the absolute temperature. It can be seen in Fig. 5(d), the FIR of the $I_{756 \text{ nm}}/I_{540 \text{ nm}}$ regularly changes with the temperature increasing. It shows a nonlinear variation of FIR value at the temperature range 303 to 573 K. The eqn (2) is used to well fit the data, the C_1 , C_2 , C_3 , C_4 and ΔE are 0.15, -0.3, 2.98, -6.1 and 429.15 cm⁻¹, respectively.

In the experiment, absolute sensitivity (S_a) is a non-negligible evaluation index of temperature, which indicates the absolute change of FIR in unit temperature, as shown in eqn (3).

$$S_{\rm a} = \left| \frac{\rm dFIR}{\rm d} \right| \tag{3}$$

Relative sensitivity (S_r) is also an important parameter that can be described as:

$$S_{\rm r} = \left| \frac{1}{\rm FIR} \frac{\rm dFIR}{\rm d} \right| \tag{4}$$

 S_a and S_r of the sample are shown in Fig. 5(e) and (f), which are fitted by eqn (3) and (4). The S_a increases with the temperature enhancement, reaching a maximum of 0.0017 K⁻¹ at 573 K. While the maximum value of S_r is 1.27% K⁻¹ at 573 K.

Fig. 6(a) shows that the emission intensities of the green and red band vary with temperature increasing. It can see that the intensity of red emission decreases more slowly than that of green emission. The energy levels 5S_2 , 5F_4 and 5F_5 energy levels are far apart, and the particle population at the two energy levels do not follow the Boltzmann-type distribution. The 5S_2 , 5F_4 and ⁵F₅ belong to non-TCLs. So the traditional FIR analysis method based on TCLs is not appropriate for the non-TCLs system. In the previous report, the relation of the non-TCLs FIR and temperature can be fitted by eqn (5).26

$$FIR_2 = \frac{I_{540 \text{ nm}}}{I_{663 \text{ nm}}} = a + bT + cT^2$$
 (5)

Fig. 6(b) shows that the original data for the $I_{540 \text{ nm}}/I_{663 \text{ nm}}$ as a function of the temperature fitting with the eqn (5). Consequently the FIR based on 5S2, 5F4/5F5 (Ho3+) shows the significant temperature dependence and well fits the eqn (5). The curve of S_a and S_r by calculation and fitting are shown in Fig. 6(c) and (d). The S_a goes down with the increasing of temperature. The maximum S_a is 0.0158 K^{-1} at 303 K. While the S_r reaches a maximum of 0.41% K⁻¹ at 393 K. The above analysis of the sensitivity of SrWO₄:Ho³⁺/

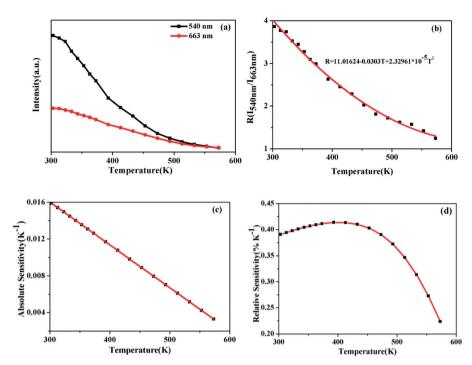


Fig. 6 (a) The emission intensity of $SrWO_4:1\%HO^{3+}/2\%Yb^{3+}$ phosphor center at red (663 nm) and green (540 nm) changes with temperature. (b) The FIR ($I_{540 \text{ nm}}/I_{663 \text{ nm}}$), (c) the absolute sensitivity and (d) the relative sensitivity of $SrWO_4:HO^{3+}/Yb^{3+}$ phosphor as a function of temperature.

Yb³⁺ based on the TCLs and the non-TCLs shows that the sample has high sensitivity and it has certain application value in optical temperature measurement.

Besides, the Er³⁺-Yb³⁺ co-doped SrWO₄ phosphor has also been studied for its sensing properties. As shown in Fig. 7(a),

the UC emission spectra of green light in the different temperature are recorded. The UC luminescence intensities of 528 nm and 550 nm (generated by ${}^2{\rm H}_{11/2} \rightarrow {}^4{\rm I}_{15/2}$ and ${}^4{\rm S}_{3/2} \rightarrow {}^4{\rm I}_{15/2}$) vary differently with the change of temperature. At 313 K, the intensity of the two emission bands is almost equal, with the

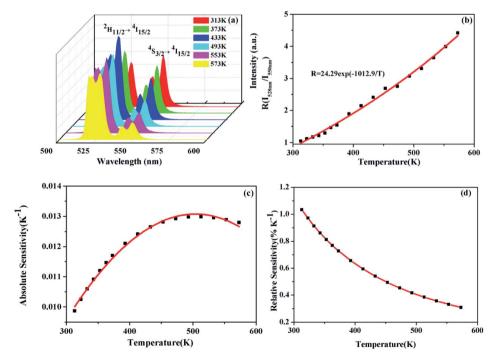


Fig. 7 (a) Temperature-dependent green UC emission spectra of $1\% Er^{3+}/1\% Yb^{3+}$ co-doped SrWO₄ phosphor at 313–573 K. (b) dependence of FIR for $I_{528 \text{ nm}}/I_{550 \text{ nm}}$ on absolute temperature. (c and d) The variation of absolute sensitivity and the relative sensitivity for SrWO₄: $1\% Er^{3+}/1\% Yb^{3+}$ phosphor with temperature between 313 K and 573 K.

Paper

0.15

• SrWO₄:Ho³⁺/Yb³⁺:I_{540 nm}/I_{663 nm}

• SrWO₄:Er³⁺/Yb³⁺:I_{528 nm}/I_{550 nm}

• SrWO₄:Er³⁺/Yb³⁺:I_{528 nm}/I_{550 nm}

• SrWO₄:Er³⁺/Yb³⁺:I_{528 nm}/I_{550 nm}

• SrWO₄:Er³⁺/Yb³⁺:I_{528 nm}/I_{550 nm}

• Temperature(K)

Fig. 8 (a) Δ FIR and (b) Δ T of SrWO₄:Ho³⁺/Yb³⁺ based on the non-TCLs (5 S₂, 5 F₄/ 5 F₅) and SrWO₄:Er³⁺/Yb³⁺ based on the TCLs (2 H_{11/2}/ 4 S_{3/2}) pair at 303–573 K.

temperature increasing, the luminescence intensity of the 528 nm emission band is gradually increased within 313–433 K, and that of the 550 nm emission band gradually decreases. Then, in the range of 433–573 K, as the temperature increasing, the intensities of two bands also decrease because of the non-radiative relaxation enhancement. But the luminescence intensity of the 550 nm emission band decreases faster than that of 528 nm emission band. When the temperature increases, the electrons at $^4\mathrm{S}_{3/2}$ state are excited. The transition from $^4\mathrm{S}_{3/2}$ to $^2\mathrm{H}_{11/2}$ occurs. It indicates that the $^2\mathrm{H}_{11/2}$ and $^4\mathrm{S}_{3/2}$ states of Er $^{3+}$ can be regarded as TCLs, and the electrons population at $^2\mathrm{H}_{11/2}$ and the $^4\mathrm{S}_{3/2}$ states follows the Boltzmann distribution. The FIR of TCLs can be expressed by the eqn (6):

$$FIR = \frac{I_{528 \text{ nm}}}{I_{550 \text{ nm}}} = C \exp\left(-\frac{\Delta E}{kT}\right)$$
 (6)

The changing of FIR of $I_{528~\rm nm}/I_{550~\rm nm}$ with temperature is displayed in Fig. 7(b), which is fitted by eqn (6). According to the fitting curve of the experimental data, the calculated C value is 24.29 and $\Delta E/k$ is 1012.9, so the ΔE between $^2H_{11/2}$ and $^4S_{3/2}$ is 703.7 cm $^{-1}$, which is very close to the experiment value of 758 cm $^{-1}$, the latter is obtained from the UC emission spectrum. Fig. 7(c) shows the S_a of SrWO₄:1%Er $^{3+}$ /1%Yb $^{3+}$

phosphor. The $S_{\rm a}$ of the phosphor increases at first and then decreases, reaching a maximum of 0.013 K⁻¹ at 513 K. In Fig. 7(d) the $S_{\rm r}$ also is calculated, and the $S_{\rm r}$ reaches a maximum of 1.03% K⁻¹ at 313 K.

In addition, thermal resolution (ΔT) is also a key parameter, which can be defined as:

$$\Delta T = \frac{\Delta \text{FIR}}{\text{FIR}} \frac{1}{S_{\text{r}}} = \frac{\Delta \text{FIR}}{S_{\text{a}}}$$
 (7)

where Δ FIR is the uncertainty of FIR, S_r is the relative sensitivity, and S_a is the absolute sensitivity. The Δ FIR can be calculated by the following equation.

$$\Delta FIR = \frac{\sum_{i=1}^{i=n} \left| FIR_i - \overline{FIR} \right|}{n}$$
 (8)

where $\overline{\text{FIR}}$ and the corresponding uncertainty ΔFIR are the average and the standard deviation values from 20 measurements at each temperature, FIR_i is the *i*-th FIR measured at the same temperature and *n* is the number of measurements.

For SrWO₄:Ho³⁺/Yb³⁺ phosphor, the thermal resolution of temperature measurement based on the non-TCLs (5S_2 , 5F_4 / 5F_5) FIR has been investigated. While the thermal resolution of temperature measurement based on the TCLs (${}^2H_{11/2}$ / ${}^4S_{3/2}$) also has been discussed for SrWO₄:Er³⁺/Yb³⁺ phosphor. The values

Table 1 Comparison of the maximum sensitivity of Ho³⁺(Er³⁺)/Yb³⁺ doped in different material

Materials	Energy level	$S_{\rm r} \left({ m K}^{-1} ight)$	$S_{\rm a}\left({ m K}^{-1} ight)$	Reference
Bi ₂ SiO ₅ :Yb ³⁺ ,Tm ³⁺	${}^{3}\mathrm{F}_{3,/2}/{}^{1}\mathrm{G}_{4}$	1.95% (300 K)	0.0168 (300 K)	7
NaGdF ₄ :Er ³⁺ /Yb ³⁺	$^{2}H_{11/2}/^{4}S_{3/2}$	1.29% (303 K)	0.0382 (363 K)	28
LaF_3 : Er^{3+}/Yb^{3+}	$^{2}H_{11/2}/^{4}S_{3/2}$	$844/T^{-2}$ (150 K)	0.0025 (400 K)	29
$Y_2O_3:Ho^{3+}/Yb^{3+}/Zn^{3+}$	${}^{5}\mathrm{F}_{3}/{}^{3}\mathrm{K}_{8}$	1067.76/T ⁻² (299 K)	0.003 (673 K)	30
$NaLa(MnO_4)_2:Sm^{3+}/Tb^{3+}$	$^{4}G_{5}/_{2}/^{5}D_{4}$	1.93 (498 K)	0.119 (265 K)	31
$BaY_2Si_3O_{10}:Ho^{3+}/Yb^{3+}$	${}^{5}F_{5}/{}^{5}S_{2}, {}^{5}F_{4}$	0.49% (298 K)	0.023 (298-448 K)	32
$BaY_{2}Si_{3}O_{10}:Er^{3+}/Yb^{3+}$	$^{4}\mathrm{F}_{9/2}/^{2}\mathrm{H}_{11/2}$	0.78% (298 K)	0.091 (298 K)	32
LaAlO ₃ :Er ³⁺ /Yb ³⁺	$^{2}\mathrm{H}_{11/2}/^{4}\mathrm{S}_{3/2}$	575.27/T ⁻² (281 K)	0.0032 (281 K)	33
$NaY(WO_4)_2:Er^{3+}/Yb^{3+}$	$^{2}H_{11/2}/^{4}S_{3/2}$	1043.12/T ⁻² (133 K)	0.0112 (515 K)	34
SrWO ₄ :Er ³⁺ /Yb ³⁺	$^{2}\mathrm{H}_{11/2}/^{4}\mathrm{S}_{3/2}$	1.03% (313 K)	0.013 (513 K)	This work
SrWO ₄ :Ho ³⁺ /Yb ³⁺	${}^{5}\mathrm{S}_{2}/{}^{5}\mathrm{F}_{4}$	1.27% (573 K)	0.0017 (573 K)	This work
SrWO ₄ :Ho ³⁺ /Yb ³⁺	${}^{5}\mathrm{S}_{2},{}^{5}\mathrm{F}_{4}/{}^{5}\mathrm{F}_{5}$	0.41% (393 K)	0.0158 (303 K)	This work

of ΔFIR and ΔT are presented among the range of 303–573 K in Fig. 8(a) and (b), respectively. It can be seen that the ΔFIR of SrWO₄:Ho³⁺/Yb³⁺ is lower than that of SrWO₄:Er³⁺/Yb³⁺. At 303 K, the ΔT of SrWO₄:Ho³⁺/Yb³⁺ and SrWO₄:Er³⁺/Yb³⁺ have the minimum values of 0.72 K and 1.14 K, respectively. The values of ΔT become increasingly with the temperature enhancement. Therefore, the samples have better accuracy at lower temperature.

In order to correctly evaluate experimental results, the other materials doped with different RE ions are compared in Table 1. It can clearly conclude that the sensitivity of our sample is higher than that of the most of samples in the table.

4. Conclusion

In conclusion, the SrWO $_4$:Ho $^{3+}$ (Er $^{3+}$)/Yb $^{3+}$ phosphors are successfully prepared by solid state reaction method. The UC luminescence properties of SrWO $_4$:Ho $^{3+}$ (Er $^{3+}$)/Yb $^{3+}$ samples are investigated under 980 nm excitation. Furthermore, the temperature sensing properties of the Ho $^{3+}$: 5S_2 / 5F_4 (TCLs), Er $^{3+}$: $^2H_{11/2}$ / $^4S_{3/2}$ (TCLs) and the Ho $^{3+}$: 5S_2 , 5F_4 / 5F_5 (non-TCLs) are investigated in the temperature range of 303–573 K. Specifically, the absolute sensitivity value reaches 0.0158 K $^{-1}$ at 303 K for Ho $^{3+}$: 5S_2 , 5F_4 / 5F_5 (non-TCLs) and the relative sensitivity value reaches 0.41% at 393 K, which is superior to most optical temperature sensing materials doped with RE $^{3+}$ ions. It means that the SrWO $_4$:Ho $^{3+}$ /Yb $^{3+}$ can achieve higher absolute sensitivity by FIR of non-TCLs, as well as it has potential application in optical temperature sensors.

Conflicts of interest

There are no conflicts to declare.

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