


 Cite this: *RSC Adv.*, 2021, **11**, 34988

# Al-doped H<sub>2</sub>TiO<sub>3</sub> ion sieve with enhanced Li<sup>+</sup> adsorption performance†

 Xianyang Dai,<sup>abc</sup> Honglong Zhan,<sup>abc</sup> Zhiqiang Qian,<sup>\*ab</sup> Jun Li,<sup>ab</sup> Zhong Liu <sup>\*ab</sup> and Zhijian Wu<sup>ab</sup>

H<sub>2</sub>TiO<sub>3</sub> (HTO) is considered to be one of the most promising adsorbents for lithium recovery from aqueous lithium resources due to its highest theoretical adsorption capacity. However, its actual adsorption capacity is much lower owing to its unknown structure and incomplete leaching of lithium. After Al is doped into H<sub>2</sub>TiO<sub>3</sub> (HTO-Al), the adsorption capacity of HTO-Al is 32.12 mg g<sup>-1</sup> and the dissolution of Ti is 2.53%. HTO-Al has good adsorption selectivity, and all the separation factors  $\alpha$  are  $\gg 1$ . Furthermore, HTO-Al also exhibits good cyclic stability and solubility resistance. After 5 cycles, the adsorption capacity remains 29.3 mg g<sup>-1</sup> and the dissolution rate is 1.7%. Therefore, HTO-Al has potential application value for recovering Li<sup>+</sup> from aqueous lithium resources.

 Received 31st August 2021  
 Accepted 18th October 2021

DOI: 10.1039/d1ra06535a

[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

## 1. Introduction

Lithium is considered as the energy metal driving progress in the world because of its many excellent properties (such as strong metallicity, high charge density and so on).<sup>1</sup> Lithium and its compounds are widely used in metallurgy, ceramics, glass, aerospace, batteries and other fields.<sup>1–8</sup> With the rapid development of the above industries, the demand for lithium has grown remarkably, while the production capacity of lithium has increased slowly. Therefore, research on lithium extraction technology is indispensable and has great application prospects. The global lithium resources mainly exist in lithium ore, salt lake brine and seawater.<sup>9</sup> In comparison, the lithium resources present in aqueous solution are more abundant, which can be exploited at a lower cost.<sup>1,10,11</sup> Hence, extracting lithium from aqueous solution is very promising and worthy of further research. The technologies of extracting lithium from aqueous solution mainly include precipitation, solvent extraction and adsorption.<sup>12–14</sup> Among these technologies, the adsorption method is fairly suitable for extracting lithium from aqueous solution and has the advantages of simple process, high recovery rate, good selectivity and environmental friendliness.<sup>15</sup>

The lithium adsorbents are mainly divided into manganese oxides, LiCl·Al(OH)<sub>3</sub>·nH<sub>2</sub>O and titanium oxides.<sup>16</sup> Although manganese oxides have been received widespread attention due to their high adsorption capacity and selectivity, the poor

solubility and cycle performance of manganese ions hinder their industrial application. In contrast, titanium oxides, especially H<sub>2</sub>TiO<sub>3</sub> (HTO), have a higher theoretic adsorption capacity and a more stable structure which are considered to be particularly promising materials. Yan *et al.* calcined lithium hydroxide and rutile type titanium dioxide, and then eluted with hydrochloric acid to obtain lithium adsorbent with a lithium adsorption capacity of 10.55 mg g<sup>-1</sup>.<sup>17</sup> Chitrakar *et al.* prepared ion sieve precursor Li<sub>2</sub>TiO<sub>3</sub> by hydrochloric acid modified high temperature solid state method, and the prepared adsorbent has a lithium absorption capacity of 4.7 mmol g<sup>-1</sup> for brine with pH = 6.5.<sup>18</sup> Gu *et al.* used CH<sub>3</sub>COOLi and TiO<sub>2</sub> as raw materials for the first time to synthesize Li<sub>2</sub>TiO<sub>3</sub> by an improved solid-state method. After pickling, H<sub>2</sub>TiO<sub>3</sub> adsorbent adsorbed West Taijinel Salt Lake, the lithium and magnesium separation factor reached 5441.17, and the adsorption capacity was 24.5 mg g<sup>-1</sup> after 5 cycles.<sup>19</sup> Zhou *et al.* prepared ion sieve precursor Li<sub>2</sub>TiO<sub>3</sub> by sol-gel method using CH<sub>3</sub>COOLi and Ti(OC<sub>4</sub>H<sub>9</sub>)<sub>4</sub> as raw materials. After elution with hydrochloric acid, the extraction rate of lithium was 50.2%, and the saturated adsorption capacity of lithium was 21.0 mg g<sup>-1</sup>.<sup>20</sup> In order to increase the adsorption capacity of Li<sup>+</sup> adsorbents, doping is usually employed as an effective method. Chitrakar *et al.* acquired Fe-doped Li<sub>4</sub>Fe<sub>x</sub>Mn<sub>5-x</sub>O<sub>12</sub>, which has lower loss rate of Mn and higher adsorption capacity of Li<sup>+</sup>.<sup>21</sup> Qian *et al.* synthesized Al-doped and Na-doped Li<sub>1.6</sub>Mn<sub>1.6</sub>O<sub>4</sub> with better solubility loss resistance and adsorption capacity.<sup>22,23</sup> For Li<sub>2</sub>TiO<sub>3</sub>, Wang *et al.* received Fe-doped and Mo-doped Li<sub>2</sub>TiO<sub>3</sub>, which possessed higher adsorption capacities.<sup>16,24</sup> Based on the above researches, it can be predicted that the doping method is an effective and available way to improve the LTO adsorption performance. Al<sup>3+</sup> own higher electronegativity than Ti<sup>4+</sup> and layered Al<sup>3+</sup> compounds (LiCl·Al(OH)<sub>3</sub>·nH<sub>2</sub>O) also have Li<sup>+</sup> adsorption performance, but there are few reports on Al-doped HTO.<sup>23,27</sup>

<sup>a</sup>Key Laboratory of Comprehensive and Highly Efficient Utilization of Salt Lake Resources, Qinghai Institute of Salt Lakes, Chinese Academy of Sciences, Xining 810008, China. E-mail: qianzq@isl.ac.cn; liuzhong@isl.ac.cn

<sup>b</sup>Key Laboratory of Salt Lake Resources Chemistry of Qinghai Province, Xining 810008, China

<sup>c</sup>University of Chinese Academy of Sciences, Beijing 100049, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d1ra06535a



In this work, Al-doped LTO with various doped amounts were prepared by improved solid-phase method. The obtained samples were characterized by XRD, SEM, TEM, XPS, FT-IR and BET, and the batch adsorption experiments (such as pH, environment temperature, lithium concentrations, adsorption selectivity and recycle stability) were systematically investigated.

## 2. Experimental

### 2.1 Preparation of Al-doped titanium lithium-ion sieves

Al-doped  $\text{Li}_2\text{TiO}_3$  (LTO) precursor was synthesized by a modified solid state reaction method according to previous literature.<sup>18</sup> In a typical procedure,  $\text{TiO}_2$ ,  $\text{LiOH} \cdot \text{H}_2\text{O}$  and  $\text{AlCl}_3 \cdot 6\text{H}_2\text{O}$  were blent and milled with anhydrous alcohol according to stoichiometric molar ratio ( $\text{Li} : \text{Ti} : \text{Al} = (2 + x) : 1 : x$ ,  $x = 2\%, 5\%, 8\%, 10\%$ ), and then dried at  $60^\circ\text{C}$ . The white mixtures were calcined at  $700^\circ\text{C}$  for 8 h at a certain heating rate and cooled in air subsequently, and then Al-doped LTO was obtained. The prepared Al-doped LTO (1 g) was treated with 200 mL  $\text{HCl}$  ( $0.2 \text{ mol L}^{-1}$ ) at room temperature under magnetic stirring for 12 h, and the Al-doped HTO adsorbent was obtained after the wash and dry process. In this work, the different Al-doped  $\text{Li}_2\text{TiO}_3$  and  $\text{H}_2\text{TiO}_3$  are designated as LTO-Al- $x$  and HTO-Al- $x$  ( $x = 2\%, 5\%, 8\%, 10\%$ ).

### 2.2 Materials characterization

The crystalline structures and morphologies of the obtained materials were determined by powder X-ray diffraction (XRD) with  $\text{Cu-K}\alpha$  radiation and scanning electron microscope (SEM, SU8000), respectively. The concentrations of  $\text{Ti}^{4+}$  and  $\text{Li}^+$  were tested by inductively coupled plasma (ICP, Optima 7000DV).  $\text{N}_2$  adsorption-desorption isotherms were used to investigate the BET surface area and pore size distributions. The surface chemical composition of doped samples was analyzed by the X-ray photoelectron spectroscopy. FT-IR spectra were collected to investigate the functional group in a range from  $400 \text{ cm}^{-1}$  to  $4000 \text{ cm}^{-1}$  on a Bruker Tensor 27 spectrometer.

### 2.3 Adsorption experiments

The batch adsorption experiments were carried out by adding 0.1 g HTO-Al-2% into 50 mL  $\text{LiCl}$  solutions ( $\text{pH} = 12$ ). The

influence factors (such as pH, temperature, initial solution concentration) on the adsorption capacity of the adsorbent were investigated. In the single factor experiments, the solution pH values were adjusted by  $\text{HCl}$  ( $0.6 \text{ mol L}^{-1}$ ) and  $\text{KOH}$  ( $10 \text{ mol L}^{-1}$ ), the initial  $\text{Li}^+$  concentration was 12, 24, 36, 48, 96  $\text{mmol L}^{-1}$ , respectively, the environmental temperature range was  $25^\circ\text{C}$ ,  $35^\circ\text{C}$ ,  $45^\circ\text{C}$ , respectively. The adsorption capacity of  $\text{Li}^+$  was calculated by eqn (1):

$$q_e = \frac{V}{m} (C_o - C_e) \quad (1)$$

where  $q_e$  represents the saturated  $\text{Li}^+$  adsorption capacity ( $\text{mg g}^{-1}$ );  $C_o$  ( $\text{mg L}^{-1}$ ) and  $C_e$  ( $\text{mg L}^{-1}$ ) are the initial and equilibrium  $\text{Li}^+$  concentration;  $m$  and  $V$  are the weight of the HTO-Al (g) and solution volume (L), respectively.

The adsorption selectivity of HTO-Al-2% was performed by adding 0.1 g adsorbents into the  $\text{MCl}$  single and mixed solution (where  $\text{M}^+ = \text{Li}^+, \text{Na}^+, \text{K}^+, \text{Rb}^+$  or  $\text{Cs}^+$ ) for 8 hours. To explore the ionic selectivity for di-valence and the practicability, the relevant practicable experiments were performed on adding 0.1 g HTO-Al-2% to the Lagoco brine (50 mL). The  $K_d$  and  $\alpha_{\text{Li M}}$  were confirmed by eqn (3) and (4):

$$K_d = (C_o - C_e) \frac{V}{mC_e} \quad (2)$$

$$\alpha_{\text{M}}^{\text{Li}} = \frac{K_{\text{Li}}}{K_d} \quad (3)$$

where  $K_d$  refers distribution coefficient,  $K_{\text{Li}}$  shows distribution coefficient of  $\text{Li}^+$ ,  $\alpha_{\text{Li M}}$  denotes separation factors.

The cycle performance of HTO-Al-2% was achieved by continuous adsorption-desorption experiments and the adsorbent was regenerated by  $0.2 \text{ mol L}^{-1}$   $\text{HCl}$  for 12 h. The contents of  $\text{Li}^+$  and  $\text{Ti}^{4+}$  were measured by ICP-OES. The dissolution of  $\text{Ti}^{4+}$  was based on eqn (4):

$$r = C_i/S \quad (4)$$

where  $r$  refers to the rate of dissolved  $\text{Li}^+$  or  $\text{Ti}^{4+}$ ;  $C_i$  represents the content of  $\text{Li}^+$  or  $\text{Ti}^{4+}$  in the solution;  $S$  is the concentration of  $\text{Li}^+$  or  $\text{Ti}^{4+}$  in the solid sample.

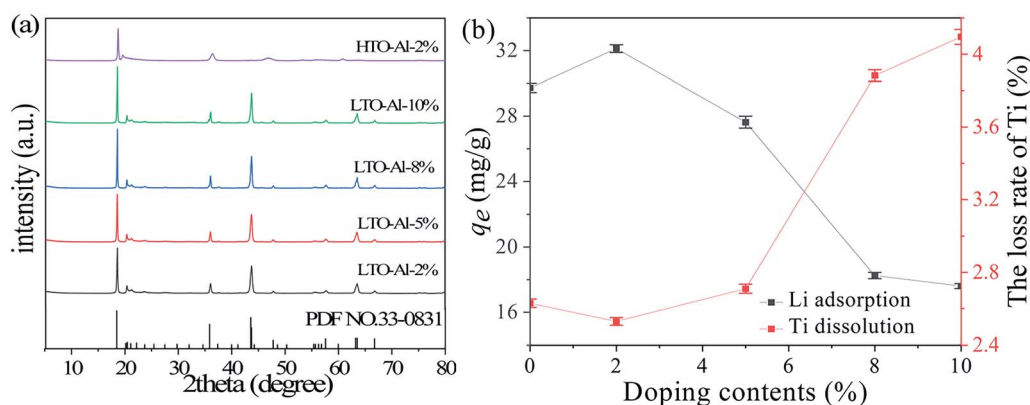


Fig. 1 (a) XRD patterns of LTO-Al- $x$  and HTO-Al-2%; (b) the effect of different Al-doped content on  $\text{Li}^+$  adsorption and Ti dissolution ( $C_{\text{Li}^+}$ :  $36 \text{ mmol L}^{-1}$ ,  $C_{\text{HCl}}$ :  $0.2 \text{ mol L}^{-1}$ ).



### 3. Results and discussion

#### 3.1 Materials characterization

Fig. 1(a) shows the XRD patterns of undoped and Al-doped precursors with Al/Ti mole ratios from 0 to 10%. All diffraction peaks of LTO-Al-*x* correspond to the layer-structured monoclinic  $\beta$ - $\text{Li}_2\text{TiO}_3$  phase (JCPDS card no. 33-0831) in *C2/c* space group.<sup>25</sup> The increase of  $\text{Al}^{3+}$  doping ratio has almost no effect on the structure of LTO precursor, which is still pure  $\text{Li}_2\text{TiO}_3$ . After pickling by HCl, the XRD patterns of HTO-Al-2% changes significantly compared with its precursor. The intensity of the diffraction peaks of HTO-Al-2% decrease obviously. The diffraction peaks of (002) and (-131) shift to the right slightly,

and (-133) peak disappears which means  $\text{Li}^+$  exchanges with  $\text{H}^+$  adequately.<sup>26</sup> Consistent with LTO, the XRD patterns of HTO have little change with the increase of  $\text{Al}^{3+}$  doping ratio (Fig. S1†). The amount of doping is an important factor which has a significant effect on the adsorption performance of the adsorbents. As shown in Fig. 1(b), as the doping amount increases, the adsorption capacities increase initially and decrease afterwards, which is contrary to the Ti dissolution. The uptake of HTO-Al-2% ( $32.12 \text{ mg g}^{-1}$ ) has the highest adsorption capacity and the Ti dissolution of HTO-Al-2% (2.5%) is slightly lower than that of undoped HTO. Hence, the doping amount of 2% is the optimal doping ratio for further investigation.

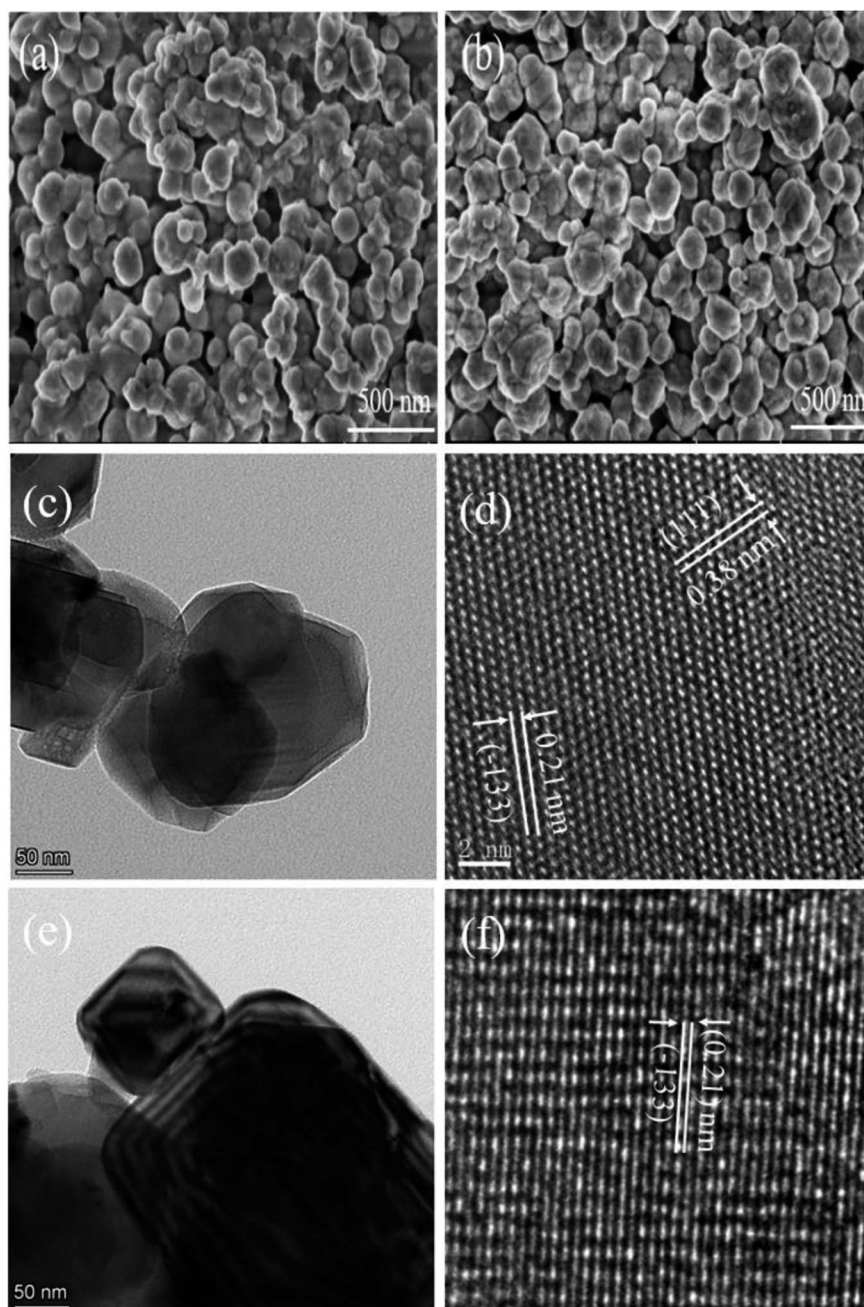


Fig. 2 SEM images of (a) LTO-Al-2% and (b) HTO-Al-2%; (c) and (e) TEM and (d) and (f) high-magnification TEM images of HTO-Al-2%.



The morphologies of LTO-Al-2% and HTO-Al-2% are revealed by SEM (Fig. 2(a) and (b)). It can be found that the pickling process has little effect on the particle morphologies and sizes of these materials. The crystal structure of HTO-Al-2% is further characterized by TEM and the results are shown in Fig. 2(c-f). The high-resolution TEM patterns of HTO-Al-2% are used to further verify the HTO-Al-2% structure (Fig. 2(d) and (f)). The lattice spacings of HTO-Al-2% are 0.21 nm and 0.38 nm, corresponding to (-133) and (111). The clear lattice fringes indicate HTO-Al-2% has well crystallinity which is consistent with XRD results.

Fig. 3 illustrates the FT-IR patterns of Al doped LTO-Al-2% and HTO-Al-2%. The peaks at  $3469.37\text{ cm}^{-1}$  and  $3477.0\text{ cm}^{-1}$  are the O-H vibrations of the adsorbed  $\text{H}_2\text{O}$ . In the spectra of HTO-Al-2%, the peaks at  $3136.65\text{ cm}^{-1}$  can be assigned to the O-H stretching vibration associated with  $\text{H}^+/\text{Li}^+$  exchange.<sup>18</sup> The band at  $636.39\text{ cm}^{-1}$  is attributed to  $\text{TiO}_6$  octahedron vibrations. Most peaks can also be observed in the spectra of HTO-Al-2%. The characteristic peaks of  $\text{TiO}_6$  both exist in LTO-Al-2% and HTO-Al-2%, which indicate that the skeleton of  $\text{TiO}_6$  octahedron is quite stable after elution with hydrochloric acid.

The XPS patterns of fitting curving for Al doped LTO and HTO are depicted in Fig. 4. The strong peaks of O 1s and Ti 2p can be easily identified, and the peak of Li 1s disappears after pickling, which is illustrated that the transformation from Li-O to H-O after the acid treatment. Fig. 4(b) shows the peak of Al 2p at  $74.2\text{ eV}$ , which is indicated that Al is doped into the adsorbent successfully and stabilized after pickling. Fig. 4(c) presents the peak of Ti 2p correspond to Ti  $2p_{1/2}$  and Ti  $2p_{3/2}$ , which is proved the chemical state of Ti is Ti(IV).<sup>27</sup> Basing on these, the valence states do not change in the adsorption process, and the  $\text{Li}^+$  adsorption behavior is an ion exchange process.<sup>28</sup> The nitrogen adsorption/desorption and the pore size distributions are revealed in Fig. S2.† The BET surface area of HTO-Al-2% is calculated to be  $12.87\text{ m}^2\text{ g}^{-1}$ . While for undoped HTO is  $4.39\text{ m}^2\text{ g}^{-1}$ . The results indicate that the BET surface area was improved by doped with Al, which may result in the increase of adsorption capacity.<sup>29,30</sup>

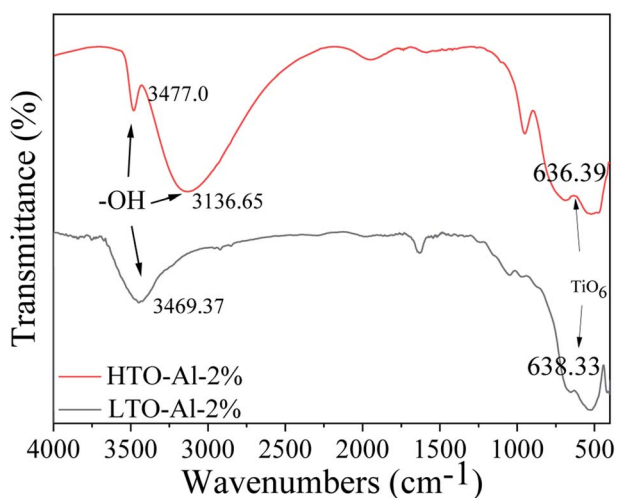


Fig. 3 The FT-IR spectra of LTO-Al-2% and HTO-Al-2%.

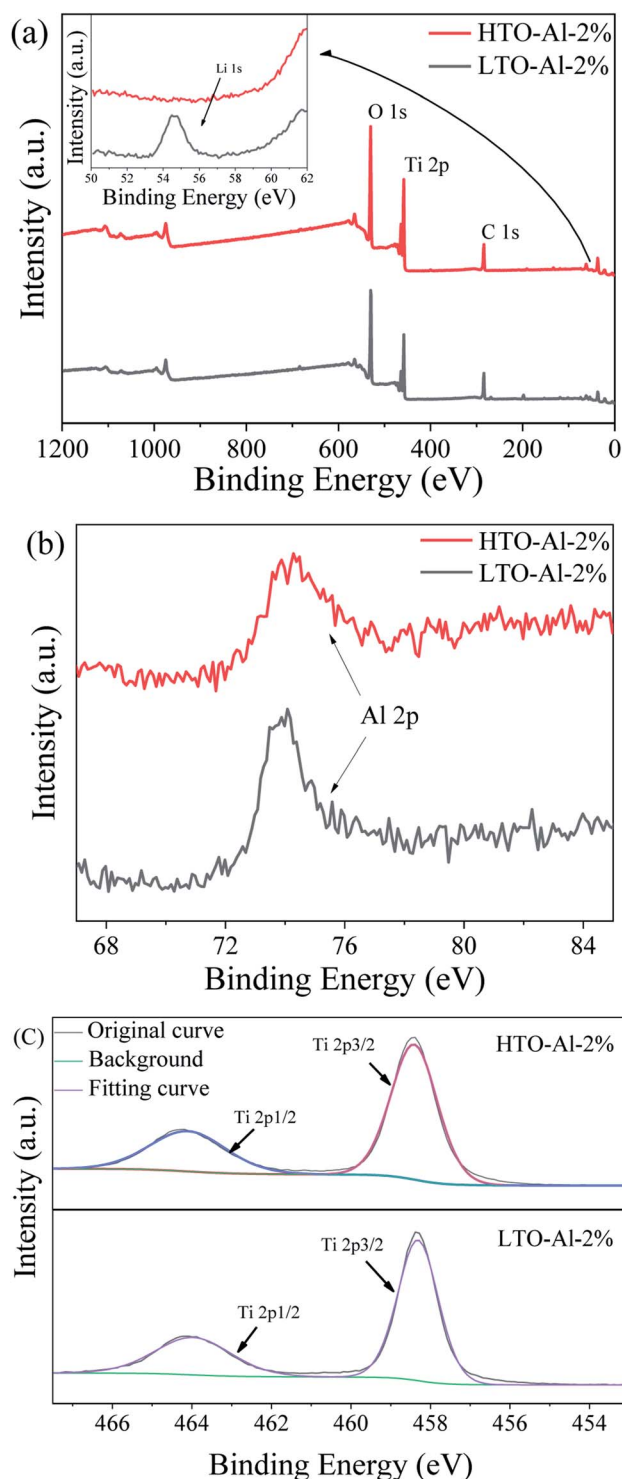


Fig. 4 XPS of LTO-Al-2% and HTO-Al-2%: (a) XPS spectra for full region; (b) XPS spectra for Al 3d; (c) XPS spectra for Ti 2p.

## 3.2 $\text{Li}^+$ adsorption

**3.2.1 Effect of pH on adsorption capacity.** As shown in Fig. 5, the pH of the solution has a significant effect on the adsorption capacity. The adsorption capacity is  $0.02\text{ mg g}^{-1}$  at  $\text{pH} = 4$  and increases slowly until  $\text{pH} = 11$ . When  $\text{pH} = 12$ , the



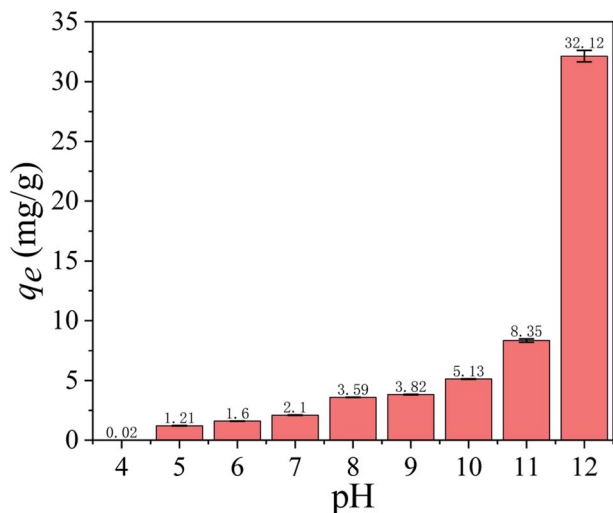
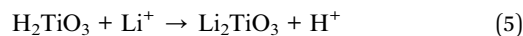


Fig. 5 Effect of pH on Li<sup>+</sup> adsorption capacity ( $C_0$ : 36 mmol L<sup>-1</sup>,  $T$ : 25 °C,  $t$ : 8 h).

adsorption capacity dramatically increases to 32.12 mg g<sup>-1</sup>. In a word, the alkaline environments are favorable for ion exchange between Li<sup>+</sup>/H<sup>+</sup>, as shown in eqn (5).



**3.2.2 Adsorption isotherms.** To illustrate the Li<sup>+</sup> adsorption behaviors, the Langmuir and Freundlich models are used to fit the adsorption process:

$$\frac{C_e}{q_e} = \frac{1}{q_m b} + \frac{C_e}{q_m} \quad (6)$$

$$\ln q_e = \ln k_F + \frac{1}{n} \ln C_e \quad (7)$$

$C_e$  (mg L<sup>-1</sup>) denotes the equilibrium Li<sup>+</sup> concentration,  $q_e$  and  $q_m$  (mg g<sup>-1</sup>) mean the empirical and theoretical maximal adsorption capacity,  $b$  indicates the Langmuir constant (L mg<sup>-1</sup>),  $k_F$  and  $n$  signify the Freundlich constant.

Table 1 The fitting results of the Langmuir and Freundlich isotherms models for Li<sup>+</sup> adsorbed on HTO-Al-2% at different reaction temperature

$T$ (°C)	Langmuir			Freundlich		
	$q_m$ (mg g <sup>-1</sup> )	$b$ (g L <sup>-1</sup> )	$R^2$	$k_F$ (mg g <sup>-1</sup> )	$1/n$	$R^2$
25 °C	34.67	0.14	0.99	22.65	0.07	0.97
35 °C	35.27	0.32	0.98	25.47	0.06	0.68
45 °C	35.67	0.38	0.99	26.74	0.05	0.66

The fitting curves are displayed in Fig. 6 and the relevant parameters are listed in Table 1. The results indicate the adsorption processes fit better by the Langmuir isotherm according to the  $R^2$ , showing that the Li<sup>+</sup> is only adsorbed in a single layer.

**3.2.3 Adsorption kinetics.** The Li<sup>+</sup> adsorption process of HTO-Al-2% was fitted by the pseudo-first-order and pseudo-second-order models for further investigation. The equations are as follows:

$$\ln(q_e - q_t) = \ln q_e - k_1 t \quad (8)$$

$$\frac{t}{q_t} = \frac{1}{k_2 q_e^2} + \frac{t}{q_e} \quad (9)$$

where  $q_e$  and  $q_t$  indicate the adsorption capacities at equilibrium or a certain time  $t$  (min),  $k_1$  and  $k_2$  denote the pseudo-first-order and pseudo-second-order coefficients.

The fitting results of the adsorption kinetics are given in Fig. 7 and Table 2. At different concentrations of Li<sup>+</sup>, all the  $R^2$  values of the pseudo-second-order model are higher than that of the pseudo-first-order model. This result means the adsorption behavior is more consistent with the pseudo-second-order model and the Li<sup>+</sup> adsorption process is a chemical adsorption.<sup>31</sup> The influence of different concentrations on the adsorption capacity of Li<sup>+</sup>, as shown in Fig. S3,<sup>†</sup> indicates that the adsorption capacity of Li<sup>+</sup> increases with the increase of the initial concentration.

**3.2.4 Adsorption selectivity.** Since coexisting ions affect the Li<sup>+</sup> adsorption process in the solutions, it is essential to

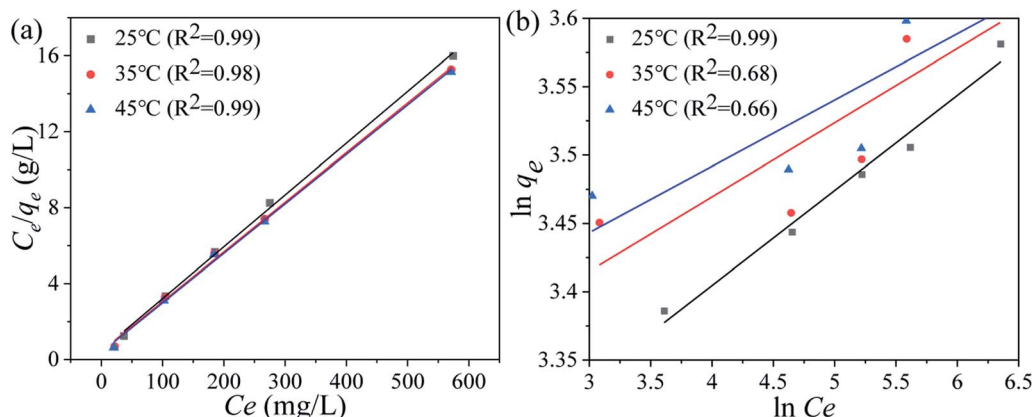


Fig. 6 (a) Langmuir and (b) Freundlich isotherm models of the Li<sup>+</sup> uptake processes.



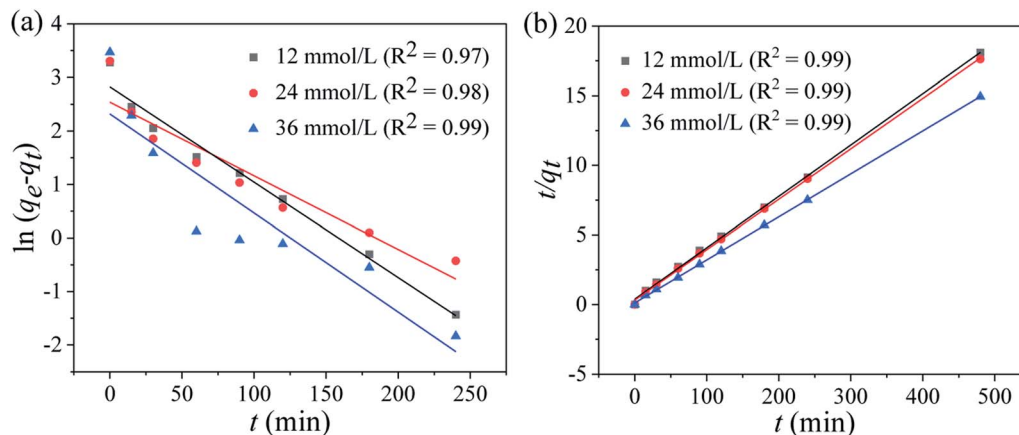


Fig. 7 (a) Pseudo-first-order and (b) pseudo-second-order kinetics models for  $\text{Li}^+$  adsorption on HTO-Al-2% at different  $\text{Li}^+$  concentrations (adsorbent: 0.1 g, volume: 50 mL, shaking speed: 180 rpm, temperature: 25 °C).

Table 2 The pseudo-first-order and pseudo-second-order kinetics models for  $\text{Li}^+$  adsorption on HTO-Al-2% at different  $\text{Li}^+$  concentrations

$C_0$ ( $\text{mg g}^{-1}$ )	Pseudo-first-order				Pseudo-second-order		
	$q_{e,\text{exp}}$ ( $\text{mg g}^{-1}$ )	$q_{e,\text{cal}}$ ( $\text{mg g}^{-1}$ )	$K_1$ ( $\text{min}^{-1}$ )	$R^2$	$q_{e,\text{cal}}$ ( $\text{mg g}^{-1}$ )	$K_1$ ( $\text{min}^{-1}$ )	$R^2$
12	26.52	25.04	0.05	0.97	27.28	0.07	0.99
24	27.23	25.63	0.06	0.98	27.53	0.1	0.99
36	32.12	31.48	0.08	0.99	33.09	0.15	0.99

investigate the ion selectivity of HTO-Al-2% in a single solution and a mixed solution. As shown in Fig. 8, the  $\text{Li}^+$  adsorption capacity in the mixed solution is 4.59  $\text{mg g}^{-1}$ , which is almost the same as that in the single  $\text{LiCl}$  solution (4.63  $\text{mg g}^{-1}$ ). Besides, the adsorption capacities of other ions are quite low in the two solutions. The distribution coefficients ( $K_d$ ) and separation factors ( $\alpha_{\text{Li M}}$ ) of various cations are shown in Table S1.† The  $\alpha_{\text{Li M}}$  of  $\text{Na}^+$ ,  $\text{K}^+$ ,  $\text{Rb}^+$ ,  $\text{Cs}^+$  are  $\gg 1$ , which is indicated HTO-

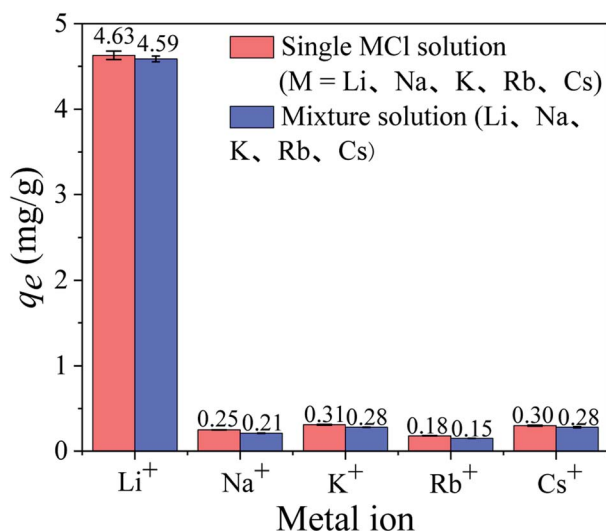


Fig. 8 The equilibrium adsorption capacities of HTO-Al-2% in the single MCl solutions and mixture solution.

Al-2% has good selectivity. This may be owing to the radius of  $\text{Li}^+$  (0.059 nm) is close to the cavities of the adsorbents while the radiuses of  $\text{Na}^+$ ,  $\text{K}^+$ ,  $\text{Rb}^+$ ,  $\text{Cs}^+$  are too large to enter the tunnel of adsorbents.

To assess the ionic selectivity for di-valence ions such as  $\text{Mg}^{2+}$  and  $\text{Ca}^{2+}$ , the relevant adsorption experiments were carried out in brine water from the Lagoco salt lake containing  $\text{Mg}^{2+}$  and  $\text{Ca}^{2+}$ . The adsorption capacities of different ion and the resulting  $K_d$  values shown in Table 3 indicate that the prepared HMO-Al-2% adsorbent possess high  $\text{Li}^+$  selectivity, implying that material has the ability to separate  $\text{Li}^+$  from solutions containing  $\text{Mg}^{2+}$  and  $\text{Ca}^{2+}$  which may have promising applications for  $\text{Li}^+$  recovery.

**3.2.5 Cycle performance.** Fig. 9 shows the  $\text{Li}^+$  adsorption uptake and loss rate of Ti in 5 cycle processes. It can be clearly seen the  $\text{Li}^+$  uptake and the loss of Ti reduced slowly in each cycle. In addition, the uptake of the fifth cycle remains 91.3% of the first time and the loss rate of Ti (1.7%) is less than 2%.

Table 3 Relevant selectivity parameters of HTO-Al-2% in Lagoco salt lake

Metal ions	$q_e$ ( $\text{mmol g}^{-1}$ )	$K_d$ ( $\text{mL g}^{-1}$ )	$\alpha_{\text{Li M}}$
$\text{Li}^+$	3.8	108.93	1
$\text{Na}^+$	0.51	0.69	157.75
$\text{K}^+$	0.17	2.61	41.71
$\text{Mg}^{2+}$	0.07	2.25	48.49
$\text{Ca}^{2+}$	0.001	5.26	20.7



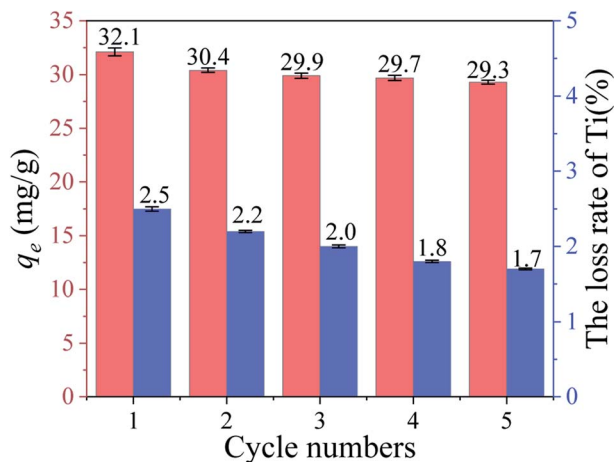


Fig. 9 The adsorption capacity and Ti dissolution of HTO-Al-2% in each regeneration process.

Compared with HTO, HTO-Al-2% reveals better adsorption property and chemical stability (Fig. S4†).

## 4. Conclusions

In this work, an effective method to improve the adsorption performances and the structure stability of the HTO adsorbent by Al doping is established. The  $\text{Li}^+$  adsorption capacity of HTO-Al ( $32.12 \text{ mg g}^{-1}$ ) is enhanced contrast with undoped HTO ( $29.73 \text{ mg g}^{-1}$ ). The adsorbents possess good  $\text{Li}^+$  selectivity in a mixed solution containing  $\text{Li}^+$ ,  $\text{Na}^+$ ,  $\text{K}^+$ ,  $\text{Rb}^+$ ,  $\text{Cs}^+$  and Lagoco salt lake. The adsorption isotherms and kinetics indicate that the  $\text{Li}^+$  adsorption behavior of HTO-Al is an endothermic process and belongs to the monolayer adsorption and chemical adsorption. In general, the HTO-Al-2% is a potential candidate as a  $\text{Li}^+$  adsorbent for industrial application.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was supported by the NSFC (No. U20A20141, No. U1607105 and No. 51302280), Thousand Talents Plan in Qinghai, Natural Science Foundation in Qinghai Province (No. 2018-GX-101, No. 2019-HZ-808 and No. 2020-ZJ-901) and Youth Innovation Promotion Association, CAS (2016377).

## References

- 1 Y. Wang, H. Liu, J. Fan, X. Liu, Y. Hu, Y. Hu, Z. Zhou and Z. Ren, *ACS Sustainable Chem. Eng.*, 2018, **7**(3), 3062–3072.
- 2 M. A. Delgado, M. C. Sánchez, C. Valencia, J. M. Franco and C. Gallegos, *Chem. Eng. Res. Des.*, 2005, **83**(9), 1085–1092.
- 3 W. Qi, J. G. Shapter, Q. Wu, T. Yin, G. Gao and D. Cui, *J. Mater. Chem. A*, 2017, **5**(37), 19521–19540.

- 4 K. Ni, X. Wang, Z. Tao, J. Yang, N. Shu, J. Ye, F. Pan, J. Xie, Z. Tan, X. Sun, J. Liu, Z. Qi, Y. Chen, X. Wu and Y. Zhu, *Adv. Mater.*, 2019, **31**(23), e1808091.
- 5 X. Cheng, F. Xian, Z. Hu, C. Wang, X. Du, H. Zhang, S. Chen, S. Dong and G. Cui, *Angew. Chem., Int. Ed. Engl.*, 2019, **58**(18), 5936–5940.
- 6 G. Zubi, R. Dufo-López, M. Carvalho and G. Pasaoglu, *Renewable Sustainable Energy Rev.*, 2018, **89**, 292–308.
- 7 L. Shan, Y. Wang, S. Liang, B. Tang, Y. Yang, Z. Wang, B. Lu and J. Zhou, *InfoMat*, 2021, **3**(9), 1028–1036.
- 8 H. Ding, J. Zhou, A. M. Rao and B. Lu, *Natl. Sci. Rev.*, 2021, **8**, nwa276.
- 9 P. W. Gruber, P. A. Medina, G. A. Keoleian, S. E. Kesler, M. P. Everson and T. J. Wallington, *J. Ind. Ecol.*, 2011, **15**(5), 760–775.
- 10 S. E. Kesler, P. W. Gruber, P. A. Medina, G. A. Keoleian, M. P. Everson and T. J. Wallington, *Ore Geol. Rev.*, 2012, **48**, 55–69.
- 11 X. Xu, Y. Chen, P. Wan, K. Gasem, K. Wang, T. He, H. Adidharma and M. Fan, *Prog. Mater. Sci.*, 2016, **84**, 276–313.
- 12 L. Gu, X.-L. Sun, W. Zhou, D.-H. Ren, D. Qiu, Z.-G. Gu and Z. Li, *J. Radioanal. Nucl. Chem.*, 2015, **307**(2), 973–983.
- 13 D. Sun, M. Meng, Y. Qiao, Y. Zhao, Y. Yan and C. Li, *Sep. Purif. Technol.*, 2018, **194**, 64–72.
- 14 H. Ge, L. Chen, W. Yuan, Y. Zhang, Q. Fan, H. Osgood, D. Matera, X.-M. Song and G. Wu, *J. Power Sources*, 2015, **297**, 436–441.
- 15 W.-J. Chung, R. E. C. Torrejos, M. J. Park, E. L. Vivas, L. A. Limjoco, C. P. Lawagon, K. J. Parohinog, S.-P. Lee, H. K. Shon, H. Kim and G. M. Nisola, *Chem. Eng. J.*, 2017, **309**, 49–62.
- 16 S. Wang, X. Chen, Y. Zhang, Y. Zhang and S. Zheng, *Particuology*, 2018, **41**, 40–47.
- 17 S. W. Yan and H. Zhong, *Ion Exch. Adsorpt.*, 1992, **8**(3), 222–228.
- 18 R. Chitrakar, Y. Makita, K. Ooi and A. Sonoda, *Dalton Trans.*, 2014, **43**(23), 8933–8939.
- 19 D. L. Gu, W. J. Sun, G. F. Han, Q. Cui and H. Y. Wang, *Chem. Eng. J.*, 2018, **350**, 474–483.
- 20 L. Y. Zhang, D. L. Zhou, Q. Q. Yao and J. B. Zhou, *Appl. Surf. Sci.*, 2016, **368**, 82–87.
- 21 R. Chitrakar, Y. Makita, K. Ooi and A. Sonoda, *Ind. Eng. Chem. Res.*, 2014, **53**(9), 3682–3688.
- 22 F. Qian, B. Zhao, M. Guo, Z. Wu, W. Zhou and Z. Liu, *Sep. Purif. Technol.*, 2021, **256**, 117583.
- 23 F. Qian, M. Guo, Z. Qian, B. Zhao, J. Li, Z. Wu and Z. Liu, *Sep. Purif. Technol.*, 2021, **264**, 118433.
- 24 S. Wang, M. Zhang, Y. Zhang, Y. Zhang, S. Qiao and S. Zheng, *Hydrometallurgy*, 2019, **187**, 30–37.
- 25 K. Kataoka, Y. Takahashi, N. Kijima, H. Nagai, J. Akimoto, Y. Idemoto and K. Ohshima, *Mater. Res. Bull.*, 2009, **44**(1), 168–172.
- 26 C. L. Yu, F. Wang, S. Y. Cao, D. P. Gao, H. B. Hui, Y. Y. Guo and D. Y. Wang, *Dalton Trans.*, 2015, **44**(35), 15721–15724.
- 27 H. Li, L. Shen, X. Zhang, J. Wang, P. Nie, Q. Che and B. Ding, *J. Power Sources*, 2013, **221**, 122–127.



## Paper

- 28 B. Zhao, M. Guo, Z. Qian, J. Li, Z. Wu and Z. Liu, *Dalton Trans.*, 2020, **49**(40), 14180–14190.
- 29 F. Qian, B. Zhao, M. Guo, Z. Qian, N. Xu, Z. Wu and Z. Liu, *Hydrometallurgy*, 2020, **193**, 105291.
- 30 F. Qian, M. Guo, Z. Qian, Q. Li, Z. Wu and Z. Liu, *Chemistryselect*, 2019, **4**(34), 10157–10163.
- 31 B. Zhao, Z. Qian, M. Guo, Z. Wu and Z. Liu, *Chem. Eng. J.*, 2021, **414**, 128729.

