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# Preparation of acid aluminum phosphate solutions for metakaolin phosphate geopolymer binder†

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This work assessed the potential of synthetic acid aluminum phosphate solutions for the enhancement of the characteristics of metakaolin phosphate geopolymer binders obtained at room temperature. The main parameters dealt with are the concentration of the initial phosphoric acid solution (40 wt%, 50 wt%, and 60 wt%) and the molar ratio Al/P (1/3 and 1.4/3) of the synthesized acid aluminum phosphate solutions. The prepared solutions have different contents and types of mono aluminum phosphate compounds (MAP) and their reactivity is pH-dependent. This is because of the continuous neutralization of the protons due to the dissolution of aluminum hydroxide, which raises the pH and decreases the conductivity. Thus the acid aluminum phosphate solutions with molar ratio Al/P of 1/3 are the most reactive. They have significantly enhanced the compressive strength of the resulting phosphate geopolymer binders. But, when compared to phosphate geopolymer obtained with pure phosphoric acid of the same concentration, the highest rate of compressive strength improvement is recorded for acid aluminum phosphate solutions having an initial concentration of phosphoric acid of 40 wt%. Thus, the modification of the composition of the phosphoric acid with the addition of the appropriate amount of aluminum is beneficial for enhancing the characteristics of phosphate geopolymer binder at any age.

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## 1. Introduction

Phosphate binders are low-carbon materials that use an acid phosphate with a metal oxide and/or aluminosilicate to develop their bonding properties.<sup>1–3</sup> They are developed mainly because of their good early properties (high early strength and bonding properties, rapid initial and final setting times).<sup>4</sup> The latter makes them suitable for application as rapid repair materials for damaged concrete structures, and waste management mainly immobilization of radioactive wastes.<sup>4</sup> The phosphate binders made of aluminosilicate as solid precursors are also named phosphate geopolymers. The replacement of metal oxide by an aluminosilicate has a positive environmental impact on the manufacturing of phosphate binder. That is because the metal oxides which occurred naturally in the form of carbonate compounds are obtained by the calcination of the respective carbonate at elevated temperature (up to 1400 °C).

The phosphate geopolymer binders show different settings and hardening behavior at an ambient or elevated temperature depending on the type of aluminosilicate used. Indeed, the reaction mechanism of aluminosilicate with and acid

phosphate include the dealumination of the aluminosilicate and the reaction of the Al<sup>3+</sup> ions with the phosphate species.<sup>5,6</sup> These reactions are very slow at room temperature and can be accelerated by applying additional heat with mild temperatures.<sup>7,8</sup> When the aluminosilicate is rich in iron, calcium, and magnesium the reaction mechanism is modified. Then during the first stage of the reaction, the dissolution process involves Al<sup>3+</sup>, Fe<sup>2+</sup>/Fe<sup>3+</sup>, Ca<sup>2+</sup>, and Mg<sup>2+</sup> ions. Afterward, those ions react preferentially with the phosphate species to form the binder as follows: Ca<sup>2+</sup> = Mg<sup>2+</sup> > Al<sup>3+</sup>, Fe<sup>2+</sup>/Fe<sup>3+</sup>.<sup>9</sup> The binder is amorphous and/or semi-crystalline depending on the synthesis conditions, and has been described as a solid solution of two or more of the following phases: aluminophosphate, silico-aluminophosphate, silicophosphate, iron phosphate, calcium phosphate, and magnesium phosphate.<sup>6,9–14</sup> In recent work, the authors demonstrated that calcium phosphate and magnesium phosphate phase are responsible for the fast hardening at room temperature of the phosphate geopolymer.<sup>9</sup> Whereas aluminum and iron phosphate phases were found responsible for the high strength development mainly at a late age. Moreover, the silicophosphate phase mainly contributes to the reinforcement of the matrix and increases the late age compressive strength of the binder.<sup>15</sup>

Metakaolin-based phosphate geopolymer binder is a slow-setting binder due to the slow rate of dissolution of aluminum in phosphoric acid. In this regard, alternative acid phosphate-containing aluminum was used to activate metakaolin, accelerate the hardening behavior, and enhance the

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Table 1 Physical and chemical composition of metakaolin

Oxides	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Fe <sub>2</sub> O <sub>3</sub>	CaO	MgO	Na <sub>2</sub> O	TiO <sub>2</sub>	K <sub>2</sub> O	MnO	P <sub>2</sub> O <sub>5</sub>	LOI	Total
MK (wt%)	56.99	35.97	1.18	0.40	0.18	0.10	1.27	0.16	0.04	0.03	1.73	98.05
Particles size distribution (μm)	<i>D</i> <sub>10</sub>				<i>D</i> <sub>50</sub>				<i>D</i> <sub>90</sub>			
	3.66				31.80				101.01			

mechanical properties. That consists mainly of the commercial monoaluminum phosphate (MAP) powder and liquid which were used to prepare MK-phosphate geopolymer. It was demonstrated that using MAP makes readily available aluminum for reacting with dissolved phosphate species at a very early stage of the reaction process. That contributes to the acceleration of the setting and hardening reactions which is most useful at room temperature.<sup>16,17</sup> However, the reactivity of the MAP depends upon its form and dosage, knowing that aluminum phosphate exists in different forms according to the molar ratio Al/P.<sup>18–20</sup>

Thus, this work aims at understanding the role of the types of monoaluminum phosphate compounds and their content for getting the optimum characteristics of the metakaolin phosphate geopolymer binder obtained at room temperature. To address this, two acid aluminum phosphate solutions were prepared by targeting specific molar concentrations Al/P that give different types and percentages of monoaluminum phosphate compounds. Then, they were used to prepare metakaolin phosphate geopolymer binders at room temperature.

## 2. Experimental methods

### 2.1 Materials

The phosphoric acid (PA) used was analytical grade ortho-phosphoric acid, 85 wt% obtained from VWR International GmbH, Germany. Analytical grade of aluminum hydroxide with formula Al<sub>2</sub>(OH)<sub>6</sub> (65% of Al<sub>2</sub>O<sub>3</sub>) and mean particle size of 40 μm was used as the aluminum source for the preparation of acid aluminum phosphate solution. Metakaolin is obtained from Argeco, France. The chemical composition of metakaolin measured by X-ray fluorescence (PW 2400 PHILIPS instrument, Eindhoven, the Netherlands) is reported in Table 1. The characteristic diameters of MK particles were measured by laser granulometry (Mastersizer 2000 from Malvern Instruments, Worcestershire, UK) and reported in Table 1.

### 2.2 Preparation of acid aluminum phosphate solution

Three concentrations of phosphoric acid solution 40 wt% (PA40), 50 wt% (PA50), and 60 wt% (PA60) were used. Two distinct amounts of aluminum hydroxide were used to get two major compositions of acid aluminum phosphate solutions with molar ratios Al/P of 1/3 and 1.4/3 each. These molar ratios were chosen to get acid aluminum phosphate solution in which the dominant components are the forms of monoaluminum phosphate (MAP) Al(H<sub>2</sub>PO<sub>4</sub>)<sub>3</sub> and/or Al(OH)(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> for the solution with molar ratio Al/P of 1/3 and 1.4/3. The role of the Al/P molar ratio on the type of monoaluminum phosphate obtained was reported in the literature.<sup>19,20</sup> Thus MAP will be used along this paper to define the acid aluminum phosphate solution prepared. The phosphoric acid was first diluted with distilled water to obtain the desired concentration, then the corresponding amount of aluminum hydroxide was added. The mix was heated at 60 °C under stirring for 10 minutes to accelerate the dissolution of aluminum hydroxide. The obtained solutions were stored for at least 24 h for allowing them to cool down before use.

### 2.3 Synthesis of MK phosphate geopolymer binder

The phosphate geopolymer binder was prepared by mixing each MAP solution with metakaolin at a constant liquid to powder ratio of 0.8. For comparison purposes, plain phosphate geopolymer was also prepared with phosphoric acid of concentrations of 40 wt%, 50 wt%, and 60 wt% using the same liquid to powder ratio of 0.8. The whole was mixed for 3 min using a kitchen aid, cast in 20 mm cubic mold later vibrated for 3 min. Afterward, all samples were covered with a plastic bag and stored in a climatic room with 20 °C and 65% relative humidity. The samples were demolded after 7 days and kept in the same condition till tests are performed. The summary of the mix composition is reported in Table 2.

### 2.4 Characterization techniques

The pH and conductivity of the different acid aluminum phosphate solutions were measured using a Mettler Toledo pH meter (SevenGo Duo Pro).

Table 2 Description of the formulations

Designation		Initial concentration of phosphoric acid (wt%)	MAP/MK mass ratio	PA/MK mass ratio
Plain phosphoric acid (PA)	MKPA40	40	—	0.8
	MKPA50	50	—	0.8
	MKPA60	60	—	0.8
Monoaluminum phosphate (MAP) Al/P = 1/3	MKMAP40	40	0.8	—
	MKMAP50	50	0.8	—
	MKMAP60	60	0.8	—
Monoaluminum phosphate (MAP) Al/P = 1.4/3	MKMAP40	40	0.8	—
	MKMAP50	50	0.8	—
	MKMAP60	60	0.8	—



The water absorption and the apparent porosity were determined according to the ASTM C-642 method, and the Archimedes principle, respectively. The samples were dried in an oven at 65 °C for 48 h and then put in water for 48 h. These tests were carried out on 28 days aged samples. The compressive strength was measured using a compression testing machine (Toni Technik, Berlin, Germany) at 7 d and 28 d. The device directly gives the value of the strength in MPa after the break is detected. The final result is the average value of 3 replicated samples for each formulation.

Attenuated Total Reflectance (ATR)-Fourier-transform infrared spectroscopy (FTIR) technique was used to collect information on the structure and composition of the samples. This was performed using Spectrum Two of the PerkinElmer instruments (UK) operating in the wavenumber range 400 to 4000  $\text{cm}^{-1}$  with a resolution of 4  $\text{cm}^{-1}$ . The changes in the mineralogy were assessed with an Empyrean PANalytical diffractometer (Malvern Panalytical Ltd, Malvern, UK) with a Ni filter transmitting the  $\text{CuK}\alpha$  radiation ( $k = 1.540598 \text{ \AA}$ ) produced by an electric current of 40 mA with a voltage of 40 kV. After being, the diffracted X-rays by the sample were recorded by a PiXcel1D detector. The thermal gravimetry analysis (TGA) coupled with differential scanning calorimetry (3+ SARE System, Mettler Toledo, Columbus, OH, USA) was used to characterize the binder. The

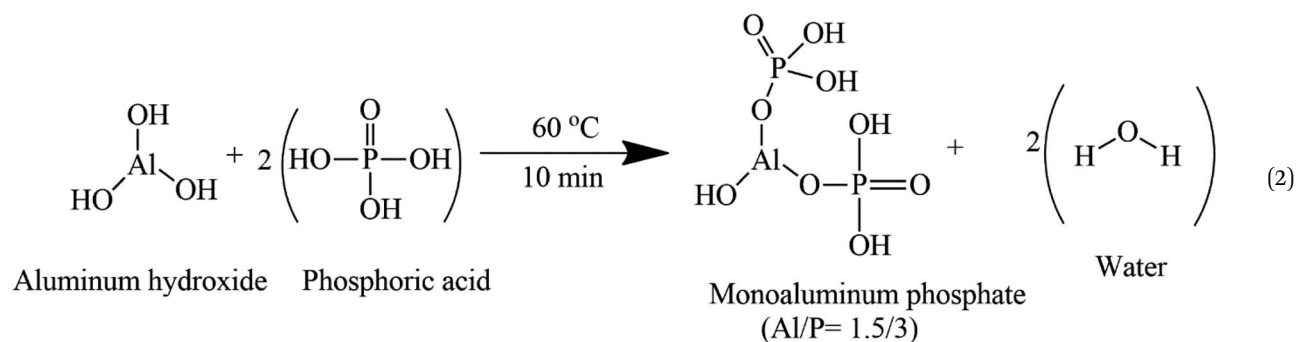
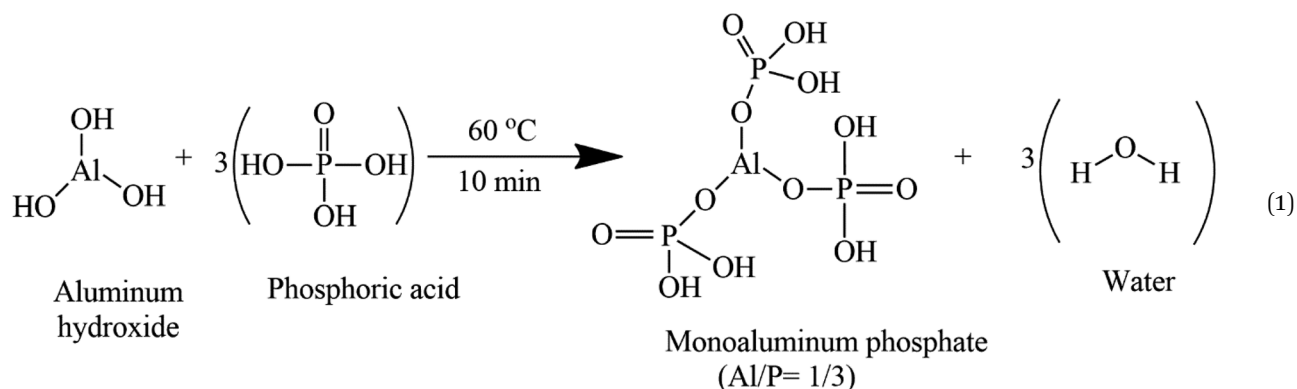
temperature range was 25–1000 °C, and it operated at a heating rate of 5  $\text{K min}^{-1}$  at a synthetic air atmosphere flowing at 70  $\text{mL min}^{-1}$ .

Pascal series mercury intrusion porosimeter (140/240, from Thermo Scientific) was used to assess the pore size distribution of the fractured samples. The microstructure of the hardened products was observed using a backscattered electron scanning electron microscope (SEM) coupled with energy-dispersive X-ray spectroscopy (EDX) (Zeiss Gemini SEM 500 NanoVP microscope Oberkochen, Germany) to determine the phase composition. The device operated in low-vacuum mode with 15 kV acceleration voltage.

### 3. Results and discussion

#### 3.1 Characterization of the acid aluminum phosphate solution

From a stoichiometric point of view, the reactions of the aluminum hydroxide with phosphoric acid using the synthesis conditions described in this paper (60 °C for 10 min) should lead to  $\text{Al}(\text{H}_2\text{PO}_4)_3$  and  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  when the molar ratio for Al/P of the initial solution is 1/3 and Al/P = 1.5/3 respectively (eqn (1) and (2)). In a recent study, Wei *et al.* reported that when an aluminum hydroxide reacts with phosphoric acid in molar ratio Al/P between 1/3 and 1.5/3 the reaction is a mixture of  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  and  $\text{Al}(\text{H}_2\text{PO}_4)_3$ . The proportion of these two



phases depends on the Al/P molar ratio. Thus, the amount of  $\text{Al}(\text{H}_2\text{PO}_4)_3$  decreases from 72.7% (Al/P = 1.1/3) to 14.3% (Al/P = 1.4/3) while the amount of  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  increases from 27.3% (Al/P = 1.1/3) to 85.7% (Al/P = 1.4/3). Thus, the acid aluminum phosphate solution obtained from this work is a monoaluminum phosphate solution with the molar ratio of Al/P 1/3 and 1.4/3, composed of a single phase of  $\text{Al}(\text{H}_2\text{PO}_4)_3$  and a mixture of  $\text{Al}(\text{H}_2\text{PO}_4)_3/\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  respectively.

**3.1.1 pH and conductivity of the acid aluminum phosphate solutions.** The acidity and the mobility of dissociated products in the different monoaluminum phosphate solutions prepared were measured to assess their potential for reacting with an aluminosilicate. Fig. 1 shows the pH and conductivity of MAP with molar ratio Al/P = 1/3 and 1.4/3. All the solutions present an acidic character with pH ranging from 0.605 to 0.985 and 1.505 to 1.778 for MAP solution with molar ratio Al/P = 1/3 and 1.4/3 respectively. The pH increases with the concentration of phosphorus in MAP and the molar ratio Al/P while conductivity follows the reverse trend. The latter varies from  $13.30 \text{ mS cm}^{-1}$  to  $4.36 \text{ mS cm}^{-1}$  and  $4.28 \text{ mS cm}^{-1}$  to  $1.26 \text{ mS cm}^{-1}$  for MAP solution with molar ratio Al/P = 1/3 and 1.4/3 respectively. It is worth noting that the pH normally increases when the concentration of proton  $\text{H}^+$  decreases. So, the behavior observed here translates that the higher is the aluminum hydroxide content and phosphoric acid concentration, the higher is the amount of  $\text{H}^+$  neutralized. This stresses that the acid-base reaction leading to the neutralization of  $\text{H}^+$  is fostered by the rate of the dissolution of aluminum hydroxide in phosphoric acid. That is supported by the fact that when aluminum hydroxide reacts with phosphoric acid in a molar ratio of  $1/3 < \text{Al/P} < 1.5/3$ , one molecule of aluminum hydroxide will preferentially react with two molecules of phosphoric acid to give  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$ . While the rest of the excess phosphoric acid will further react by partially dissolving  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  to form  $\text{Al}(\text{H}_2\text{PO}_4)_3$ .<sup>20</sup> This decreases the acidity of the MAP solution. That also justifies the decrease of the conductivity with the

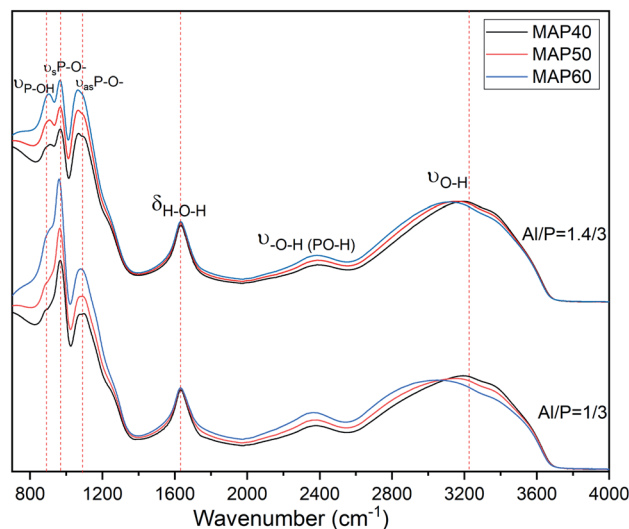


Fig. 2 ATR-FTIR spectra of monoaluminum phosphate solution with varying phosphate concentration and Al/P molar ratio.

increase in the concentration of phosphorus in MAP and the molar ratio Al/P. It was also demonstrated that the viscosity of such MAP solutions increases with the phosphoric acid concentration and aluminum dosage (Al/P molar ratio).<sup>19,20</sup> Consequently it limits the mobility of ions in the solution and reduces the conductivity. The pH rise can also be explained by the neutralization reaction occurring between aluminum ions and phosphate species in the MAP solution. Wagh reported that because of that neutralization, the solubility of aluminum decreases with the increase of the pH of the acidic solution till reaches the near-neutral pH where it starts to increase again.<sup>21</sup> So it can be summarized that within the pH range of the MAP solution prepared, the series of MAP solutions with molar ratio Al/P = 1/3 are more reactive with an aluminosilicate than MAP solution having molar ratio Al/P = 1.4/3.

**3.1.2 ATR-Fourier transformed infrared spectroscopy.** The infrared spectra of MAP solutions are presented in Fig. 2. The bands centered at  $1626\text{--}1630 \text{ cm}^{-1}$ ,  $3026\text{--}3182 \text{ cm}^{-1}$  correspond respectively to the bending vibration of the H-O-H bond and stretching vibration of -O-H bond from water molecules. The latter shifts slightly to lower wavenumbers with the increase of the concentration of the phosphorus in the solution. That is ascribed to the decrease of the protonation degree, thus the decrease of the acidity.<sup>18</sup> The band at near  $2370 \text{ cm}^{-1}$  corresponds to the intermolecular hydrogen bonding due to the interactions of the O-H bond between two molecules of aluminum phosphate. The main bands of the MAP appear at near  $880\text{--}894 \text{ cm}^{-1}$ ,  $958\text{--}964 \text{ cm}^{-1}$ ,  $1071\text{--}1079 \text{ cm}^{-1}$  for MAP with molar ratio Al/P = 1/3 and near  $896\text{--}902 \text{ cm}^{-1}$ ,  $962\text{--}964 \text{ cm}^{-1}$ ,  $1059\text{--}1063 \text{ cm}^{-1}$  for MAP with molar ratio Al/P = 1.4/3. These bands are characteristic of the various stretching vibration of the -P-O- bonds. The intensity and shape of these bands change with the phosphorus concentration and Al/P molar ratio. The intensity of the main band  $\nu_{\text{s}}\text{-P-O-}$  at near  $1059\text{--}1079 \text{ cm}^{-1}$  in all spectra becomes weak with the increase

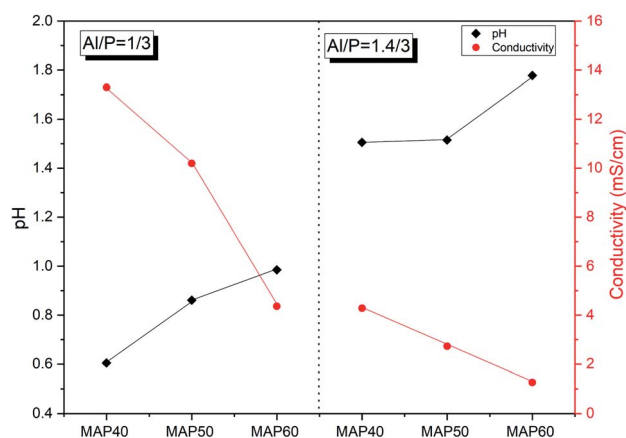


Fig. 1 pH and conductivity of monoaluminum phosphate solution with varying phosphate concentration and Al/P molar ratio.



of the molar ratio Al/P. That is due to the reaction between phosphoric acid and aluminum hydroxide to form the Al–O–P bond of the MAP molecules. As observed in the graph, the weak band of –P–OH appearing as a shoulder in the spectra of MAP obtained with molar ratio Al/P = 1/3 becomes strong and well resolved in the spectra of MAP obtained with molar ratio Al/P = 1.4/3. The band  $\nu_{\text{as}}\text{P–O–}$  at near 958–964  $\text{cm}^{-1}$  in all spectra also follow that trend as it becomes strong with the increase of the molar ratio Al/P. On the other hand, the intensity of all these bands increases with the concentration of phosphorus in MAP. That behavior is also observed in the IR spectra (Fig. S1 ESI†) of the plain phosphoric acid with the increase of concentration where their bandwidth is narrower than in the MAP solutions. The increase of the intensity and the bandwidth of the vibration of the –P–O– bond means that the amount of these bonds increases in the MAP solution. Likely due to the formation of the  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  and  $\text{Al}(\text{H}_2\text{PO}_4)_3$ . That result agrees with the literature and demonstrates that the degree of acidity of the MAP solutions decreases with the increase of the phosphorus concentration and Al/P molar ratio.<sup>19,20</sup>

### 3.2 Physical and mechanical properties

**3.2.1 Water absorption and apparent porosity.** The water absorption and apparent porosity presented in Fig. 3 vary respectively from 12.71% to 9.17% and 16.25% to 11.80% for MK phosphate geopolymer obtained with MAP molar ratio Al/P = 1/3. Moreover, for MK phosphate geopolymer obtained with MAP molar ratio Al/P = 1.4/3 the water absorption varies from 13.25% to 23.54% while the apparent porosity changes from 16.25% to 28.47%. It is observed that the water absorption and apparent porosity of phosphate geopolymer obtained with MAP molar ratio Al/P = 1/3 are lower than for those obtained with MAP molar ratio Al/P = 1.4/3. That is an indication that reaction products formed in the two systems have different microstructures. Thus, the lower porosity obtained when the MAP solution has a molar ratio Al/P = 1/3 refers to a denser microstructure.

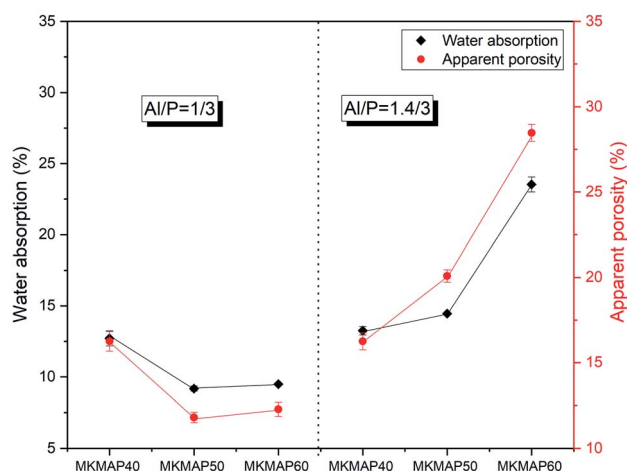


Fig. 3 Water absorption and apparent porosity of the MK phosphate geopolymer binder obtained with MAP having molar ratio Al/P = 1/3 (left) and 1.4/3 (right).

When examining the evolution of these properties with the concentration of phosphorus in the solution, one can see an inconsistency in the trend of the water absorption and apparent porosity of the two series of formulations. In the series of formulation obtained with MAP having the molar ratio Al/P = 1/3 the samples, MKMAP50 have the lowest water absorption (9.18%) and apparent porosity (11.80%). Whereas in others with MAP having the molar ratio Al/P = 1.4/3 both water absorption and apparent porosity increase with the concentration of phosphorus. That lack of consistency can be explained by the different types of molecules of aluminum phosphate contained in the MAP solution and their polymerization degree that lead to a phosphate geopolymer with reaction products having different chemical compositions. That will be discussed in detail later on in this paper.

**3.2.2 Compressive strength evolution.** The compressive strength evolution at 7 and 28 days presented in Fig. 4 shows that the strength of phosphate geopolymer obtained with MAP having molar ratio Al/P = 1/3 is higher than those obtained with MAP molar ratio Al/P = 1.4/3. That agrees with the results of the evolution of the pH and conductivity (Fig. 1) which demonstrated that the MAP solutions with molar ratio Al/P = 1/3 are more reactive and suitable to effectively dissolve the aluminosilicate than MAP solutions having molar ratio Al/P = 1.4/3. Therefore, one can assume that the higher rate of dissolution–condensation of metakaolin when reacted with MAP solutions having molar ratio Al/P = 1/3 has fostered the formation of a high volume of reaction products. This has given rise to phosphate geopolymer samples with fewer voids (low porosity). This also confirms the lower water absorption and apparent porosity of phosphate geopolymer obtained with MAP molar ratio Al/P = 1/3 compared to those obtained with MAP having molar ratio Al/P = 1.4/3.

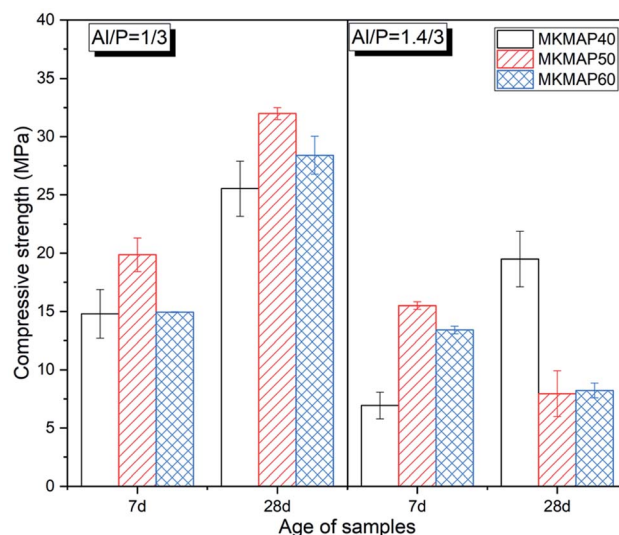


Fig. 4 Compressive strength evolution of the MK-based phosphate geopolymer binder obtained with MAP having molar ratio Al/P = 1/3 (left) and 1.4/3 (right).

The compressive strength ranges from 14.79 MPa to 19.87 MPa at 7 d and 25.53 MPa to 31.98 MPa at 28 d for MK phosphate geopolymer binder obtained from MAP solution with molar ratio  $\text{Al/P} = 1/3$ . The optimum compressive strength in this series was achieved for MKMAP50 at all ages. It must be recalled that the reaction products are formed when the amount of dissolved species reaches the critical value to undergo condensation. In other words, there is a window of the  $\text{Al/P}$  and  $\text{Si/P}$  molar ratio of the phosphate binder within which the physical properties are the best.<sup>11,22–24</sup> The 7 d compressive strength of the series of samples obtained with MAP having molar ratio  $\text{Al/P} = 1.4/3$  is the highest (15.5 MPa) for MKMAP50. At 28 d the compressive strength of samples MKMAP40 in that series becomes the highest one while the compressive strength of samples MKMAP50 and MKMAP60 dropped to 7.95 MPa and 8.23 MPa, respectively. The decrease of the compressive strength with time means that the binder or the reaction products in those samples are not stable and tend to self-deteriorate upon aging. This has also been observed in the literature.<sup>11</sup> The significant increase of the compressive strength of MKMAP40 (obtained with MAP having molar ratio  $\text{Al/P} = 1.4/3$ ) with time indicates that there is a continuous dissolution of metakaolin with the time that contributes to the formation of a high volume of reaction products leading to a sample with reducing porosity as shown in Fig. 3.

To assess the suitability of the acid aluminum phosphate solutions to react with metakaolin in comparison to the diluted phosphoric acid solution, the strength activity index (SAI) was measured. The SAI is used very often to assess the strength improvement in blended Portland cement due to the pozzolanic activity of the added pozzolanic materials.<sup>25</sup> The SAI is defined in this work as the ratio of the compressive strength (28 d) of each MKMAP binder to one of its correspondent MKPA obtained with similar phosphoric acid solutions (PA40, PA50, and PA60) at the same age. The results of the measured strength activity index are presented in Fig. 5. It must be noted that the

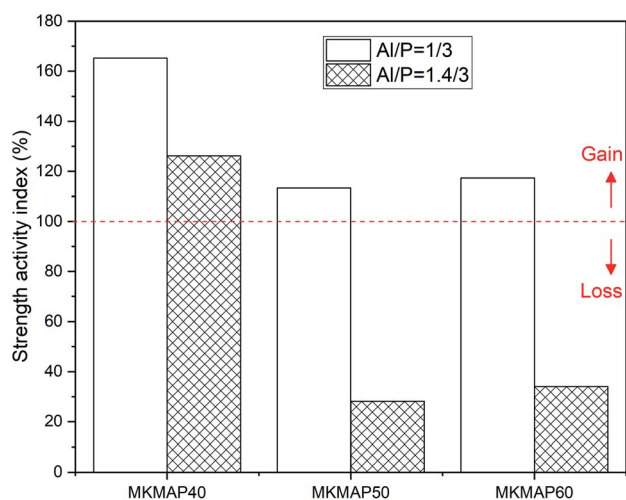


Fig. 5 Effect of phosphorus concentration and molar ratio  $\text{Al/P}$  of the MAP solution on the strength activity index.

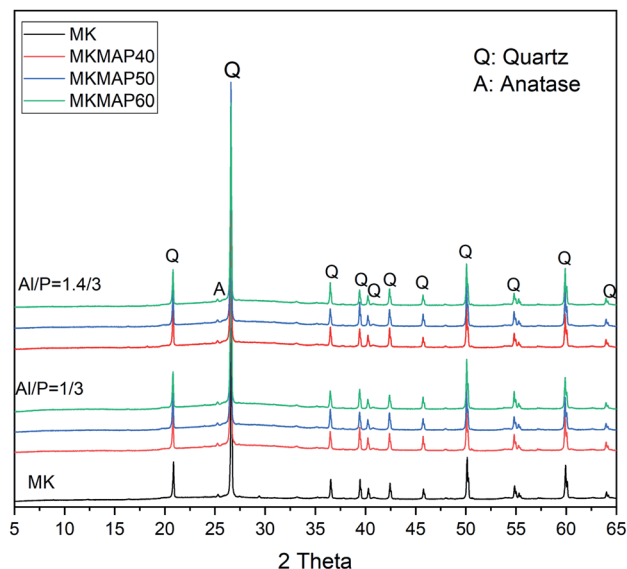


Fig. 6 X-ray patterns of MK and MKMAP samples prepared from different concentrations of phosphorus and molar ratio  $\text{Al/P}$ .

red dotted line is the reference beyond which the gain in compressive strength was achieved, whereas below the line it is the loss in strength. Therefore one can observe that the SAI is higher for the phosphate geopolymer binder obtained with MAP having a molar ratio  $\text{Al/P} = 1/3$ . Also, their SAI is higher than 100%, which means that the use of MAP solutions having molar ratio  $\text{Al/P} = 1/3$  for the preparation of MK phosphate geopolymer binder is beneficial for strength improvement. That corroborates the previous statement made in this work stressing that the ready availability of soluble aluminum species in the right amount in the phosphate solution accelerates the reaction. When considering the series of formulations obtained with MAP solutions having molar ratio  $\text{Al/P} = 1.4/3$ , only the

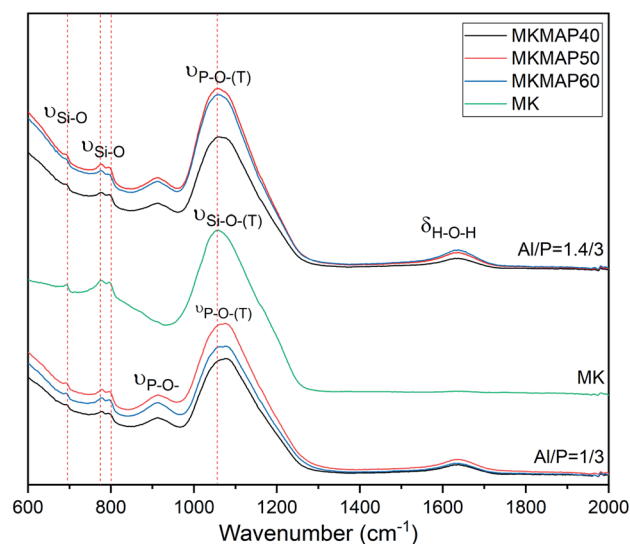


Fig. 7 ATR-FTIR spectra of MK and MKMAP samples prepared from different concentrations of phosphorus and molar ratio  $\text{Al/P}$ .

sample MKMAP40 has the SAI that crosses the reference line (100%). This means that the other samples MKMAP50 and MKMAP60 lose strength at 28 d compared to the respective ones obtained from PA50 (MKPA50) and PA60 (MKPA60). In other words, the amount of aluminum hydroxide added to these solutions (PA50 and PA60) to get the final Al/P molar ratio of 1.4/3, delays the reaction process and is detrimental to the compressive strength development at a late age. This corroborates the recent findings where the addition of aluminum only improved the early age compressive strength and decreased the late age strength.<sup>17</sup>

Regardless of the type of MAP solution, the samples MKMAP40 have the highest SAI. The 28 d compressive strength of the series of samples obtained with different concentrations of pure phosphoric acid (Fig. S2 ESI†) is lower for MKPA40 (15.46 MPa) than the one of MKPA50 (28.23 MPa) which is also higher than the one of MKPA60 (24.20 MPa). Moreover, within each series of MAP solutions, one can see that the SAI of the resulting phosphate binder decreases with the increase of the concentration of the initial phosphoric acid used. All this indicates that the availability of soluble aluminum at the early stage of the reaction is most effective in accelerating the

reaction kinetic and strength development of low concentration and pH of acid aluminum phosphate solution. Therefore, this ascertains the benefit of using soluble aluminum to compensate for its low availability when preparing phosphate geopolymer from low soluble phosphoric acid solution to improving its compressive strength.

### 3.3 Microstructural characteristic of the phosphate geopolymer binder

**3.3.1 Mineralogy and structural phase composition.** The mineralogical phases of the binder presented in Fig. 6 show that all the mineral phases present in MK are still present in the reacted products. These phases are quartz as a major one and anatase. No significant change could be identified in the phosphate binders with the variation of the concentrations of phosphorus and the molar ratio Al/P of the MAP solution. That is obvious as silica which is the main constituent of quartz is sparsely soluble in the pH range of the acid phosphate solution.<sup>9,15</sup>

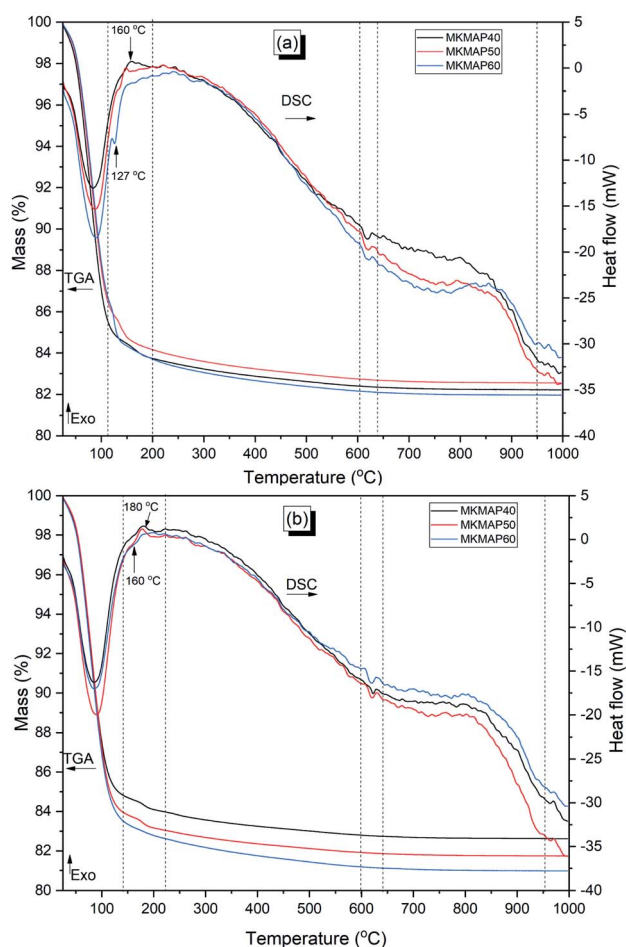


Fig. 8 TGA-DSC of MK-based phosphate geopolymer binder obtained with MAP having molar ratio Al/P = 1/3 (a) and 1.4/3 (b).

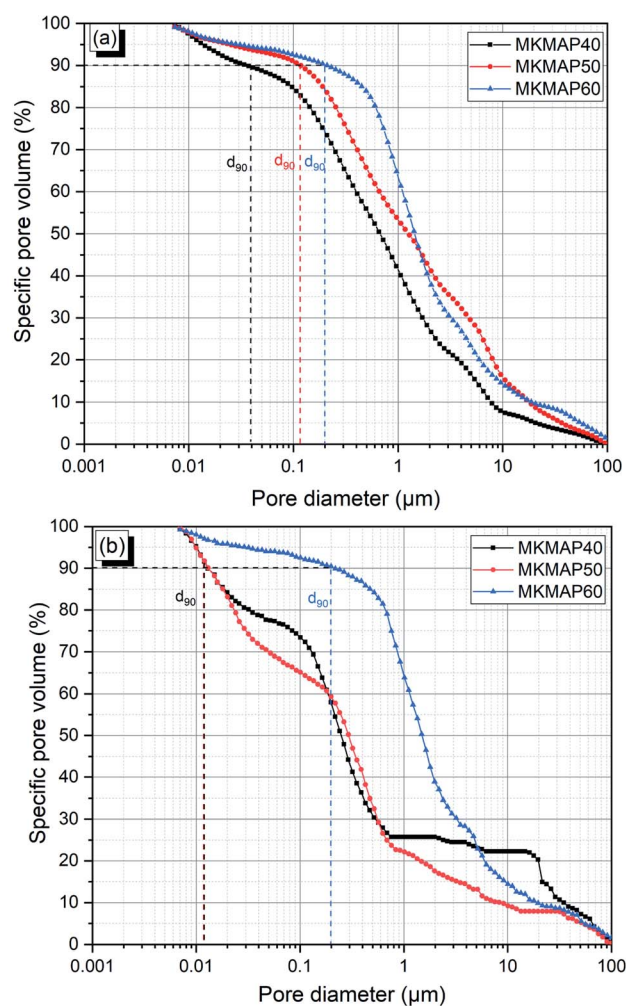


Fig. 9 Pore size distribution of the phosphate geopolymer binder obtained with: (a) MAP solution having molar ratio Al/P = 1/3 and (b) MAP solution having molar ratio Al/P = 1.4/3.





More information on the structural compositions and the resulting changes with the concentrations of phosphorus and molar ratio Al/P of the MAP can be seen in the ATR-FTIR in the range  $600\text{--}2000\text{ cm}^{-1}$  of the phosphate binder showed in Fig. 7. One can see in the spectra the bands appearing at near  $692\text{ cm}^{-1}$ ,  $776\text{ cm}^{-1}$  and  $798\text{ cm}^{-1}$  characteristics of the Si-O stretching vibration from quartz.<sup>26,27</sup> The band at  $1630\text{ cm}^{-1}$  present in all spectra except MK is characteristic of the bending vibration of the H-O-H bonds of the water. In the spectrum of MK, the main band is broad and centered at  $1056\text{ cm}^{-1}$ . It corresponds to the stretching vibration of the Si-O-(T) bond where T is either Si or Al. Such broadband in an IR spectrum indicates the availability of several types of disordered bonds that come mainly from the amorphous phase of MK. In the spectra of MKMAP, that broadband has shifted to a higher wavenumber and is centered at  $1077\text{ cm}^{-1}$  and  $106\text{ cm}^{-1}$  for MKMAP obtained with molar ratio Al/P = 1/3 and 1.4/3 respectively. That main band characteristic of the phosphate binder corresponds to the stretching vibration of the P-O-(T)

bond where T is Al and/or Si. So, such broadband indicates the presence of P-O-T vibration bonds corresponding to various types of aluminum phosphate/silicophosphate phases and is characteristic of the amorphous phase of the binder. The increase of the wavenumber is ascribed to an increase in the degree of condensation-polycondensation of the dissolved species, which leads to a longer polymeric chain with stronger bonds Si-O-P-O-Al/Al-O-P.<sup>14,22</sup> The shift of the wavenumber of the main band is more pronounced ( $21\text{ cm}^{-1}$ ) for the series of MKMAP obtained with MAP solution having molar ratio Al/P = 1/3 than the other series obtained with MAP having molar ratio Al/P = 1.4/3 ( $11\text{ cm}^{-1}$ ). The latter corroborates well with the previous results presented in this work which show that MAP solutions with molar ratio Al/P = 1/3 are more reactive than MAP with molar ratio Al/P = 1.4/3. Another change observed involves the presence of the band at  $911\text{ cm}^{-1}$  not present in the spectrum of MK and characteristic of the stretching vibration of the P-O- bond of the phosphate binder.<sup>28</sup>

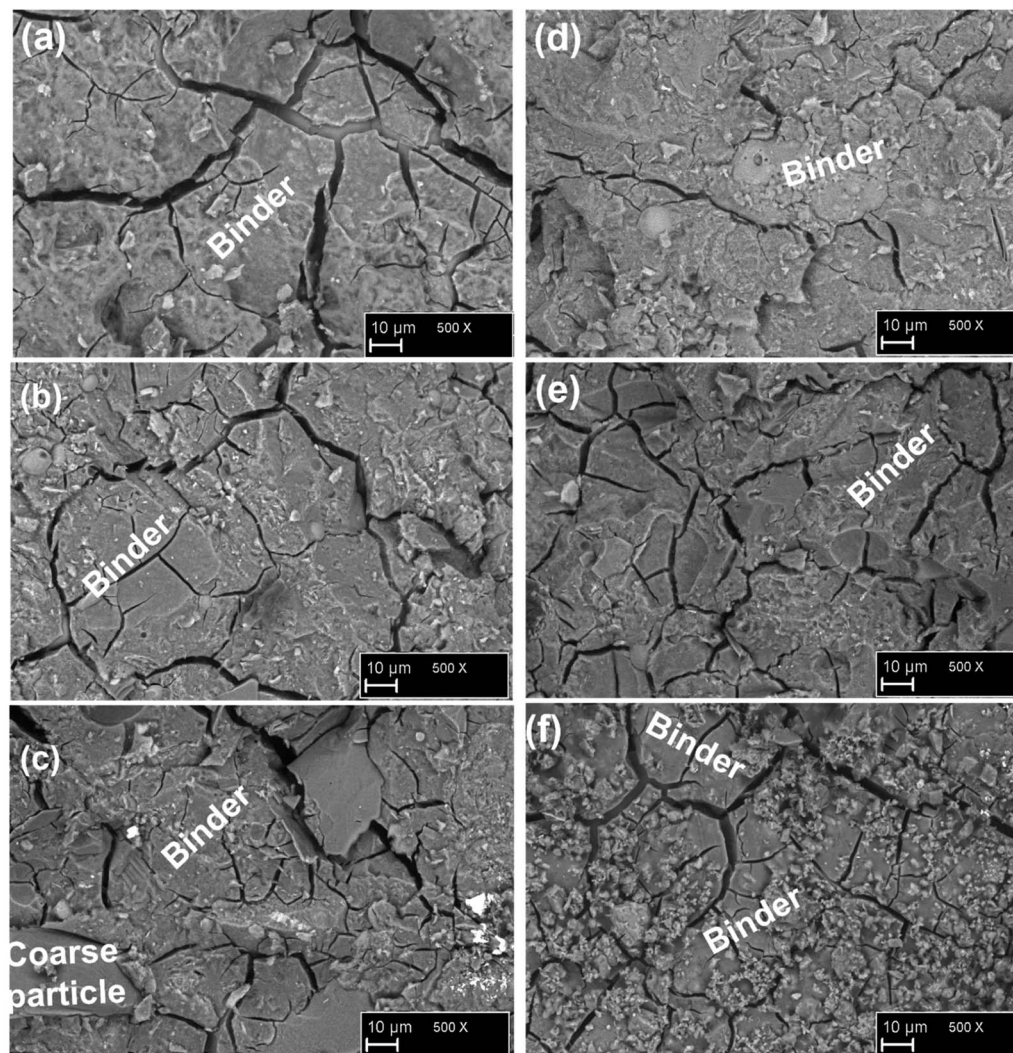
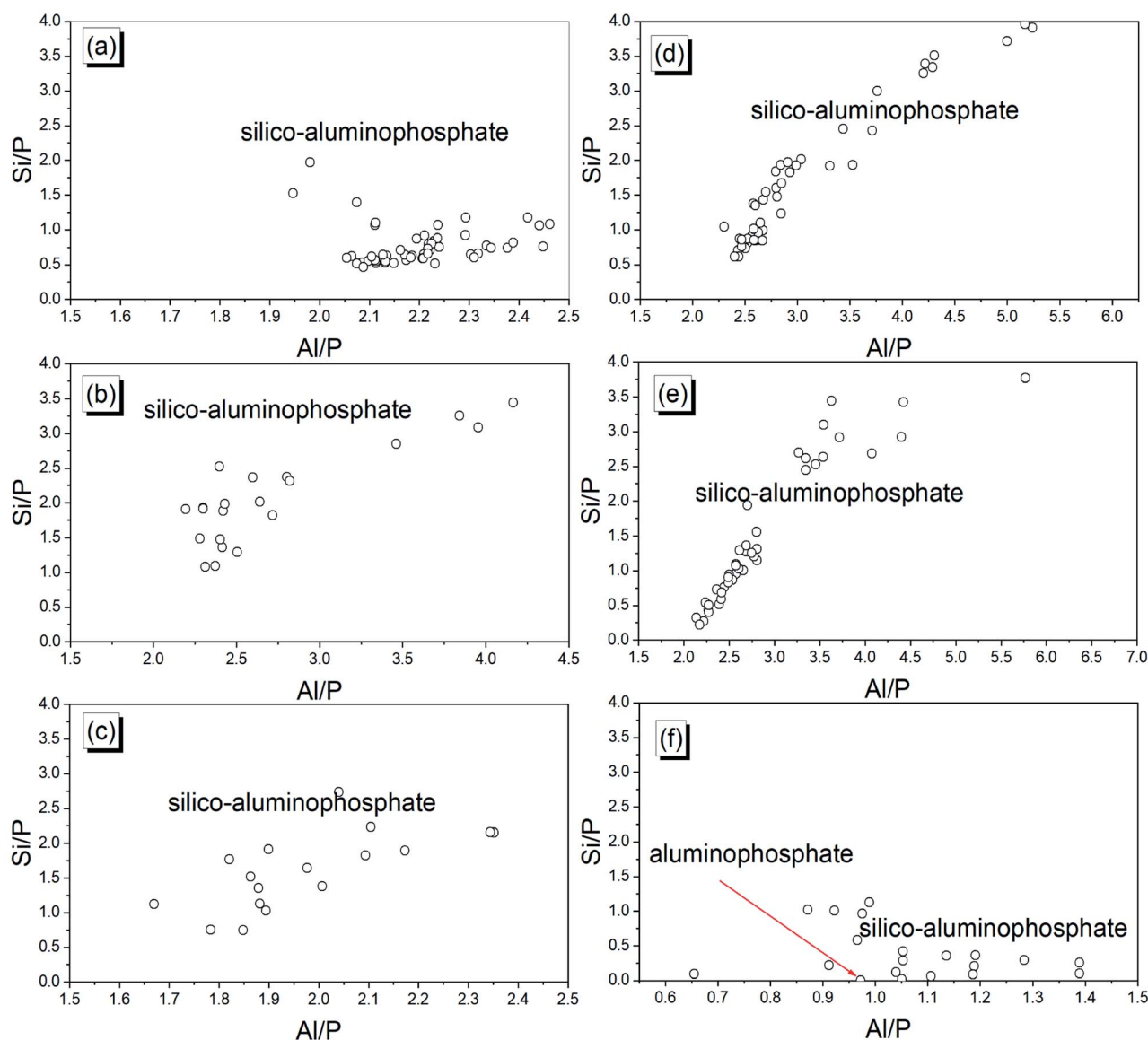


Fig. 10 BSE micrographs of MK-based phosphate geopolymer binder obtained with: (a)–(c) MAP having molar ratio Al/P = 1/3 (MKMAP40 (a), MKMAP50 (b) and MKMAP60 (c)); (d)–(f) MAP having molar ratio Al/P = 1.4/3 (MKMAP40 (d), MKMAP50 (e) and MKMAP60 (f)).



**3.3.2 Phases stability and transition.** Fig. 8 depicts the TGA-DSC curves of the phosphate geopolimer binders. Before 100 °C there is a major endothermic peak accompanied by a mass loss which is characteristic of the evaporation of the physical bond water. As the temperature rises there is a mass loss accompanied by an endothermic peak appearing at near 127 °C and 160 °C on the MKMAP obtained with MAP having molar ratio Al/P = 1/3 and 1.4/3 respectively. It is characteristic of the removal of the chemically bonded water of the aluminum phosphate hydrated of the binder into aluminum phosphate ( $\text{AlPO}_4$ ).<sup>29,30</sup> This, later on, gives rise to trigonal aluminum phosphate ( $\alpha\text{-AlPO}_4$ ) characterized by the exothermic peak at 146–160 °C (Fig. 8a) and 174–180 °C (Fig. 8b) with no noticeable mass loss. That exothermic peak (160 °C and 180 °C) is well-featured for MKMAP40 and MKMAP50. It is not identifiable

for MKMAP60 obtained with MAP having molar ratio Al/P = 1/3, while it appears as a very weak peak on MKMAP60 obtained with MAP having molar ratio Al/P = 1.4/3. This is related to the content of the amorphous aluminum phosphate phase being converted into crystallized aluminum phosphate ( $\alpha\text{-AlPO}_4$ ). Another exothermic feature is detected at 795–854 °C (Fig. 8a) and 805–818 °C (Fig. 8b), and corresponds to the polymorphic transformation of the trigonal aluminum phosphate ( $\alpha\text{-AlPO}_4$ ) phase into tetragonal aluminum phosphate ( $\beta\text{-AlPO}_4$ ).<sup>30</sup> The last exothermic peak appears at 970 °C and is characteristic of the crystallization into mullite of the unreacted MK after acid phosphate activation. The crystallization of mullite is also observed in the DSC curves of the MK where it is more pronounced than in the phosphate binder (Fig. S3 ESI†). Moreover, it is noticeable an endothermic peak at 620 °C in all



**Fig. 11** Atomic ratios Si/P vs. Al/P of MK-phosphate geopolimer binder: (a)–(c) obtained with MAP having molar ratio Al/P = 1/3 (MKMAP40 (a), MKMAP50 (b) and MKMAP60 (c)); (d)–(f) obtained with MAP having molar ratio Al/P = 1.4/3 (MKMAP40 (d), MKMAP50 (e) and MKMAP60 (f)).



binders and MK, which is accompanied by a mass loss well featured in the TGA of MK (Fig. S3 ESI†). That corresponds to the dehydroxylation of the residual kaolinite and indicates that the calcination process was not efficient.

### 3.4 Microstructure

The pore size distribution determined from MIP is shown in Fig. 9. Roughly, the diameter of the pores ranges from 0.007  $\mu\text{m}$  to 100  $\mu\text{m}$  in all samples. On the other hand, the multiple slope change as highlighted in all curves of the specific pore volume *vs.* pore diameter demonstrates that all the mixes have a multimodal pore distribution.<sup>31</sup> That indicates the heterogeneity of the pores structures of the phosphate binder developed. Within each series, the characteristic pore diameter  $d_{90}$  (corresponding to the maximum size of 90% of the pores present in the phosphate binder) is 0.03  $\mu\text{m}$  (MKMAP40), 0.118  $\mu\text{m}$  (MKMAP50), and 0.21  $\mu\text{m}$  MKMAP60 for the MKMAP obtained with MAP solution having molar ratio Al/P = 1/3. Whereas in the other series prepared with the MAP solution having molar ratio Al/P = 1.4/3,  $d_{90}$  is 0.013  $\mu\text{m}$  for MKMAP40 and MKMAP50 and 0.20  $\mu\text{m}$  for MKMAP60. These results indicate that the porosity is dominated by the capillary pores, since  $d_{90}$  ranges between 0.01 to 1  $\mu\text{m}$ , which corresponds to a more connected pore network. A clear trend of the pores size distribution could not be observed which confirms the complexity of the pore structure of the binder developed. Whilst from the Archimedes method, the trend of the apparent porosity measured correlates well with the compressive strength evolution. The discrepancies of the MIP result with the Archimedes method rely on the fact that MIP does not measure internal pore and the porosity is a function of the pressure applied, which always gives different results.<sup>31</sup> But it was used here to assess the pore size distribution of the binders.

The distribution of different phases of the phosphate geopolymer binders at the microstructural level is depicted in Fig. 10. The micrographs on the fractured samples are characterized by microcracks and coarse particles of various sizes. Moreover, dense and homogeneous areas are also visible in all samples which may be ascribed to the binding phases resulting from the acid–base reaction between MAP and MK. The different phases are shapeless, meaning that the reaction products are amorphous. This correlates well with the XRD results where no new mineral was detected. The result of the EDX analysis performed on at least 20 points of the reacted phases is shown in Fig. 11. The dense and homogeneous phases appearing as the major phase in all samples are identified as silico-aluminophosphate. Another phase identified by EDX as the aluminophosphate phase (Fig. 11f) has a gel-like structure and is stuck on the surface of the dense phase. The silico-aluminophosphate phase is simply pure silica diffused in an aluminophosphate phase or the mixture of silicophosphate/aluminophosphate. That was confirmed by the DSC curves and also agrees with literature that showed the crystallization of the aluminophosphate phase indicating that it stands as the major phase beside others in the phosphate geopolymer binder.<sup>32,33</sup> When looking closely at the range of atomic ratio Si/P, one can distinguish two types of the composition of the silico-aluminophosphate phase based on the silicon content. Those with the atomic ratios Si/P < 0.5 imply a silica poor silico-

aluminophosphate phase, whereas the atomic ratios Si/P > 0.5 correspond to the silica-rich silico-aluminophosphate phase. The poor silica-based silico-aluminophosphate phase is the consequence of the low rate of dissolution of MK as discussed previously.

## 4. Conclusion

This work reported the role of the concentration of phosphorus and molar ratio Al/P of synthesized acid aluminum phosphate solution in its suitability for the activation of metakaolin at room temperature. The main findings are summarized as follows:

(1) The characterization of the acid aluminum phosphate solution revealed that it is mainly composed of various types of monoaluminum phosphate (MAP) compounds depending on the aluminum dosage. For solutions with molar ratio Al/P = 1/3, the main compound is  $\text{Al}(\text{H}_2\text{PO}_4)_3$ , while those with molar ratio Al/P = 1.4/3  $\text{Al}(\text{OH})(\text{H}_2\text{PO}_4)_2$  as the main compound and  $\text{Al}(\text{H}_2\text{PO}_4)_3$  as the secondary.

(2) The acid aluminum phosphate solutions obtained with molar ratio Al/P = 1/3 are most reactive because of their higher acidity as compared to the solutions with molar ratio Al/P = 1.4/3.

(3) The highest 7 and 28 d compressive strength was achieved by the phosphate geopolymer binders obtained with solutions having molar Al/P = 1/3 and the initial concentration of phosphoric acid 50 wt%. That is because of the availability of the appropriate amount of Al and P in the matrix that leads to the strongest binder.

(4) The early availability of soluble aluminum from the MAP is beneficial for accelerating the reaction kinetic and the compressive strength of the resulting phosphate geopolymer as compared to those prepared with diluted phosphoric acid only. Further, this is most effective when the appropriate amount of aluminum has been used.

## Author contributions

Jean Noël Yankwa Djobo, design, carried out the experiments, and wrote the manuscript, Rachel Yanou Nkwaju performed some experiments and revised the final manuscript.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 A. S. Wagh, in *Advances in ceramic matrix composites*, ed. N. P. Bansal, J. P. Singh and W. M. Kriven, The American Ceramic Society, 2005, pp. 107–116.
- 2 A. S. Wagh, in *Developments in Strategie Materials and Computational Design II Developments in Strategie Materials and Computational Design II*, ed. W. M. Kriven, A. L. Gyekenyesi and J. Wang, John Wiley & Sons, Inc., Hoboken, New Jersey, 2011, pp. 91–103.
- 3 A. S. Wagh and S. Y. Jeong, *J. Am. Ceram. Soc.*, 2003, **44**, 1838–1844.
- 4 A. S. Wagh, *Chemically Bonded Phosphate Ceramics Twenty-First Century Materials with Diverse Applications*, Elsevier Science, Amsterdam, 2 edn, 2016, vol. 24.
- 5 A. Katsik, *Adv. Appl. Ceram.*, 2019, 1–13.
- 6 Y.-S. Wang, Y. Alrefaei and J.-G. Dai, *Front. Mater.*, 2019, **6**, 1–17.
- 7 M. Zribi, B. Samet and S. Baklouti, *J. Non-Cryst. Solids*, 2019, **511**, 62–67.
- 8 H. Celerier, J. Jouin, N. Tessier-Doyen and S. Rossignol, *J. Non-Cryst. Solids*, 2018, **500**, 493–501.
- 9 J. N. Y. Djobo, D. Stephan and A. Elimbi, *J. Build. Eng.*, 2020, **31**, 101427.
- 10 M. Lassinantti Gualtieri, M. Romagnoli and A. F. Gualtieri, *J. Eur. Ceram. Soc.*, 2015, **35**, 3167–3178.
- 11 A. Katsiki, T. Hertel, T. Tysmans, Y. Pontikes and H. Rahier, *Materials*, 2019, **12**, 1–15.
- 12 T. Dong, S. Xie, J. Wang, Z. Chen and Q. Liu, *J. Aust. Ceram. Soc.*, 2020, **56**, 175–184.
- 13 C. R. Kaze, G. L. Lecomte-Nana, E. Kamseu, P. S. Camacho, A. S. Yorkshire, J. L. Provis, M. Duttine, A. Wattiaux and U. C. Melo, *Cem. Concr. Res.*, 2021, **140**, 106320.
- 14 J. N. Y. Djobo, A. Elimbi and D. Stephan, *SN Appl. Sci.*, 2020, **2**, 828.
- 15 B. Zhang, H. Guo, P. Yuan, L. Deng, X. Zhong, Y. Li, Q. Wang and D. Liu, *Cem. Concr. Compos.*, 2020, **110**, 103601.
- 16 Y.-S. Wang, J.-G. Dai, Z. Ding and W.-T. Xu, *Mater. Lett.*, 2017, **190**, 209–212.
- 17 Y.-S. Wang, J. L. Provis and J. G. Dai, *Cem. Concr. Compos.*, 2018, **93**, 186–195.
- 18 E. Palacios, P. Leret, M. J. De La Mata, J. F. Fernández, A. H. De Aza and M. A. Rodríguez, *CrystEngComm*, 2013, **15**, 3359–3365.
- 19 Y. Li, G. Chen, S. Zhu, H. Li, Z. Ma, Y. Liu and L. Liu, *Bull. Mater. Sci.*, 2019, **45**(5), DOI: 10.1007/s12034-019-1912-3.
- 20 H. Wei, T. Wang, Q. Zhang, Y. Jiang and C. Mo, *J. Chin. Chem. Soc.*, 2020, **67**, 116–124.
- 21 A. S. Wagh, *Chemically Bonded Phosphate Ceramics*, 2016, pp. 141–155.
- 22 H. K. Tchakouté, C. H. Rüschler, E. Kamseu, F. Andreola and C. Leonelli, *Appl. Clay Sci.*, 2017, **147**, 184–194.
- 23 H. Douiri, S. Louati, S. Baklouti, M. Arous and Z. Fakhfakh, *Mater. Lett.*, 2014, **116**, 9–12.
- 24 S. Louati, S. Baklouti and B. Samet, *Adv. Mater. Sci. Eng.*, 2016, **2016**, DOI: 10.1155/2016/2359759.
- 25 S. Kumar, J. N. Y. Djobo, A. Kumar and S. Kumar, *Eur. J. Environ. Civ. Eng.*, 2020, **24**, 833–848.
- 26 C. N. Bewa, H. K. Tchakouté, C. Banenzoué, L. Cakanou, T. T. Mbakop, E. Kamseu and C. H. Rüschler, *Appl. Clay Sci.*, 2020, **198**, 105813.
- 27 B. Zhang, H. Guo, L. Deng, W. Fan, T. Yu and Q. Wang, *Appl. Clay Sci.*, 2020, **199**, 105887.
- 28 S. Louati, W. Hajjaji, S. Baklouti and B. Samet, *Appl. Clay Sci.*, 2014, **101**, 60–67.
- 29 P. L. Zhuravleva, N. S. Kitaeva, Y. M. Shiryakina and A. A. Novikova, *Russ. J. Appl. Chem.*, 2016, **89**, 367–373.
- 30 P. Prado-Herrero, J. Garcia-Guinea, E. Crespo-Feo and V. Correcher, *Phase Transitions*, 2010, **83**, 440–449.
- 31 H. Giesche, *Part. Part. Syst. Charact.*, 2006, **23**, 9–19.
- 32 M. Zribi and S. Baklouti, *Polym. Bull.*, 2021, DOI: 10.1007/s00289-021-03829-0.
- 33 H. Guo, B. Zhang, L. Deng, P. Yuan, M. Li and Q. Wang, *Appl. Clay Sci.*, 2021, **204**, 106019.

