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## Efficient visible-light photocatalytic H<sub>2</sub> evolution with heterostructured Ag<sub>2</sub>S modified CdS nanowires†

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The low separation efficiency of photogenerated charges and severe photocorrosion seriously impeded the application of CdS in photocatalytic water splitting. Here we report new routes to improve the photocatalytic performance of CdS nanowires (NWs) by decorating with Ag<sub>2</sub>S nanoparticles, so Ag<sub>2</sub>S/CdS heterojunction is constructed. The Ag<sub>2</sub>S/CdS heterojunction exhibited optimal photocatalytic H<sub>2</sub> evolution rate of 777.3 μmol h<sup>-1</sup> g<sup>-1</sup>, which is 12.1 times higher than that of pure CdS. The intrinsic characteristics of Ag<sub>2</sub>S/CdS nanocomposites, such as structure, optical properties, and surface chemical state are systematically studied by experimental characterizations and theoretical calculations. The comprehensive analysis demonstrates that the heterojunction between Ag<sub>2</sub>S and CdS accelerates photoinduced electrons transfer from CdS to Ag<sub>2</sub>S, enhancing their ability for water splitting. Meanwhile, the holes on the valence band of CdS react with the sacrificial agents, thus leading to the efficient separation of photogenerated electron–hole pairs. This work offers a simple route to synthesize one-dimensional CdS-based nanocomposites for efficient energy conversion driven by visible light.

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### 1. Introduction

Water splitting for hydrogen evolution by semiconductor photocatalysts is one of the most promising solutions to alleviate the energy crisis. To improve solar-to-hydrogen energy conversion efficiency, numerous photocatalysts, such as TiO<sub>2</sub>,<sup>1</sup> g-C<sub>3</sub>N<sub>4</sub>,<sup>2</sup> CdS,<sup>3</sup> ZnS,<sup>4</sup> SrTiO<sub>3</sub>,<sup>5</sup> and so on, have been intensively studied over the last few decades. Especially, CdS is widely considered to be the most promising candidate in photocatalytic hydrogen evolution reaction (HER) due to its outstanding properties such as the proper band edge locations, moderate bandgap, and strong visible light absorption.

However, the high recombination rate of photogenerated charges and severe photocorrosion seriously hinder the application of CdS. Therefore, it is urgent to improve the electron–hole separation rate and enhance the photocatalytic stability for CdS. In recent years, one-dimensional (1D) nanostructured materials are extensively constructed because they have high aspect ratios which are beneficial for H<sub>2</sub> evolution. It's well-known that 1D nanomaterials can accelerate the long-distance

charge transfer velocity since 1D constructure provides a direct charge transport path along the axial direction for direct conduction.<sup>6</sup> In addition, the 1D materials can also enhance light-harvesting capacity and scattering properties which are substantially important for hydrogen evolution. As one of the outstanding examples of the 1D catalysts, CdS nanowires (NWs) have been successfully synthesized and used for photocatalytic HER. Nevertheless, the photocatalytic activity of CdS NWs is still unsatisfactory because of their poor reusability and depressed hydrogen production. Recently, it was demonstrated that the photocatalytic performance can be substantially improved by combining CdS with other cocatalysts including metal/alloy, metal phosphides, metal oxide/hydroxide, carbide, perovskite nanocrystals, metal sulfides, metal organic frame and so on. Especially, transition metal sulfides have attracted great attention owing to their unique optical, electrical, electrochemical, and other properties. For instance, 1D/2D CdS/CoS<sub>x</sub> heterojunction was found to be very efficient for the hydrogen evolution;<sup>7</sup> CdS–MnS composite had very high stability which can maintain its catalytic ability for over 42 h of measurement;<sup>8</sup> MoS<sub>2</sub> NSs–CdS NWs system also possessed high visible-light H<sub>2</sub> evolution rate that far exceeded pure CdS NWs.<sup>9</sup>

As a typical transition metal sulfide, Ag<sub>2</sub>S, with admirable photosensitive and low bandgap (0.9–1.2 eV),<sup>10</sup> was extensively used in binary or ternary photocatalysts, such as rGO/Ag<sub>2</sub>S/g-C<sub>3</sub>N<sub>4</sub>,<sup>11</sup> CdS/Ag<sub>2</sub>S/g-C<sub>3</sub>N<sub>4</sub>,<sup>12</sup> Ag<sub>2</sub>S/ZnS,<sup>13</sup> and MoS<sub>2</sub>/Ag<sub>2</sub>S/Ag.<sup>14</sup> In addition, Ag<sub>2</sub>S/CdS structure was also demonstrated to possess superior photocatalytic degradation performance under visible

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light irradiation.<sup>15</sup> However, there are few reports on studying the influence of  $\text{Ag}_2\text{S}$  on  $\text{CdS}$  NWs in the photocatalytic hydrogen evolution.

In this work, we studied the photocatalytic HER activity of  $\text{CdS}$  NWs modified by  $\text{Ag}_2\text{S}$  nanoparticles (NPs) *via* a simple *in situ* ion exchange method. It was found that the 5.0%- $\text{Ag}_2\text{S}/\text{CdS}$  exhibited optimal photocatalytic  $\text{H}_2$  evolution rate of 777.3  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 12.1 times higher than that of pure  $\text{CdS}$  NWs. The results exhibited that a heterojunction is formed at the interface between  $\text{Ag}_2\text{S}$  NPs and  $\text{CdS}$  NWs, and the co-catalyzed  $\text{Ag}_2\text{S}$  NPs provide active sites and trap photo-generated electrons, which results in efficient separation of "electron-hole" pairs. In addition, theoretical calculations based on density functional theory (DFT) were carried out to uncover the behind mechanism for the enhancement of photocatalytic activity. Our work demonstrates that  $\text{Ag}_2\text{S}$  NPs can be served as outstanding cocatalysts in the field photocatalytic HER from water splitting.

## 2. Experimental section

### 2.1. Chemicals

All chemical reagents are analytical grade and purchased from Aladdin Chemical Reagent Co., Ltd, AR. The ultrapure water ( $\geq 18 \text{ M}\Omega \text{ cm}@25^\circ\text{C}$ ) used in the whole experiment was purified from tap water in the laboratory.

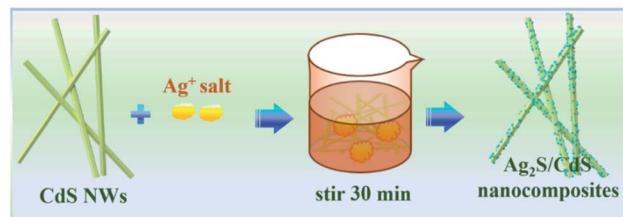
### 2.2. Photocatalyst synthesis

**(a) Synthesis of  $\text{CdS}$  NWs.**  $\text{CdS}$  NWs were prepared by a facile solvothermal method which was reported in our previous work.<sup>16</sup>

**(b) Synthesis of  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites.** The synthetic routes of  $\text{Ag}_2\text{S}$  NPs/ $\text{CdS}$  NWs nanocomposites are illustrated in Scheme 1. The  $\text{Ag}_2\text{S}/\text{CdS}$  composites were formed by the ion exchange between the  $\text{CdS}$  precursor and  $\text{Ag}^+$ . Typically,  $\text{CdS}$  NWs (1.445 g) were dispersed into 30 mL ultrapure water and then stirred for 30 min. After that, a certain amount of  $\text{AgNO}_3$  was dissolved into the  $\text{CdS}$  NWs dispersion and vigorously stirred for another 30 min. The  $\text{Ag}_2\text{S}$  nanoparticles were *in situ* formed on the  $\text{CdS}$  surface, which can avoid the quick nucleation and growth of  $\text{Ag}_2\text{S}$  in the solution. It is expected that there are two main chemical reactions existing during the phase and morphological evolution, which are given as follows:



As given in eqn (1), there is a small amount of  $\text{S}^{2-}$  in the reaction solution around the  $\text{CdS}$  nanowires, due to the dynamic equilibrium. When  $\text{AgNO}_3$  was added in the suspension, nanosized  $\text{Ag}_2\text{S}$  particles were preferentially formed and deposited on the  $\text{CdS}$  surface by the heterogeneous nucleation (eqn (2)). The solubility product constant ( $K_{\text{sp}}$ ) of  $\text{CdS}$  and  $\text{Ag}_2\text{S}$  is  $8.0 \times 10^{-27}$  and  $6.3 \times 10^{-50}$ , respectively.<sup>17</sup> Such large



Scheme 1 Schematic illustration on the synthesis of  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites.

difference in solubility provides the driving force for cation exchange, forming the final  $\text{Ag}_2\text{S}/\text{CdS}$  composites.

Finally, the products were obtained by centrifugation, alternately rinsed with ultrapure water and ethanol several times, and dried at  $60^\circ\text{C}$  for overnight. For comparison, the  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites with various content of  $\text{Ag}_2\text{S}$  NPs were synthesized under the same condition. The ingredient molar ratios of  $\text{Ag}_2\text{S} : \text{CdS}$  are 0.010, 0.025, 0.050, 0.075, 0.100, and denoted as 1.0%- $\text{Ag}_2\text{S}/\text{CdS}$ , 2.5%- $\text{Ag}_2\text{S}/\text{CdS}$ , 5.0%- $\text{Ag}_2\text{S}/\text{CdS}$ , 7.5%- $\text{Ag}_2\text{S}/\text{CdS}$ , and 10.0%- $\text{Ag}_2\text{S}/\text{CdS}$ , respectively.

Details of characterization methods and parameters, photoelectrochemical parameters, photocatalytic  $\text{H}_2$  evolution evaluation experiment, and theoretical calculation are available in the ESI.<sup>†</sup>

## 3. Results and discussions

### 3.1. Photocatalytic hydrogen evolution performance

The  $\text{H}_2$  evolution performances of pure  $\text{CdS}$  and  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites were evaluated under visible-light irradiation. Fig. 1a presented the cumulative  $\text{H}_2$  production of the as-prepared samples. Intuitively, bare  $\text{CdS}$  shows a poor photocatalytic activity, which is attributed to the low separation efficiency of the photo-generated electron-hole pairs and serious photocorrosion. In contrast, the photocatalytic  $\text{H}_2$  evolution activities of  $\text{CdS}$  are remarkably improved when composited with  $\text{Ag}_2\text{S}$ . It was found that the 5.0%- $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposite can produce hydrogen of 117.0  $\mu\text{mol}$  after 3 h of irradiation, which is superior to the other samples. Fig. 1b listed the  $\text{H}_2$  generation rate of all samples. It is clearly shown that the  $\text{H}_2$  evolution rate of the  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites gradually increases with the content of  $\text{Ag}_2\text{S}$  increasing from 1.0% to 5.0%, and achieves a maximum value of 777.3  $\mu\text{mol h}^{-1} \text{g}^{-1}$  for the 5.0%- $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposite. This value is 12.1 times higher than that of pure  $\text{CdS}$ . Further increase of the  $\text{Ag}_2\text{S}$  content leads to the decrement of  $\text{H}_2$  generation rate, which indicates that moderate content of  $\text{Ag}_2\text{S}$  in  $\text{Ag}_2\text{S}/\text{CdS}$  is needed for high performance of  $\text{H}_2$  generation. Although  $\text{Ag}_2\text{S}$  NPs offer abundant surface hydrogen evolution active sites, excessive  $\text{Ag}_2\text{S}$  cause severe aggregation and shielding effect for light absorption, which may result in the depression of photocatalytic activity.

### 3.2. Structural analysis

In order to understand the influence of the structure of the catalysts on their photocatalytic HER performance, X-ray diffractometer (XRD) measurements were performed to detect



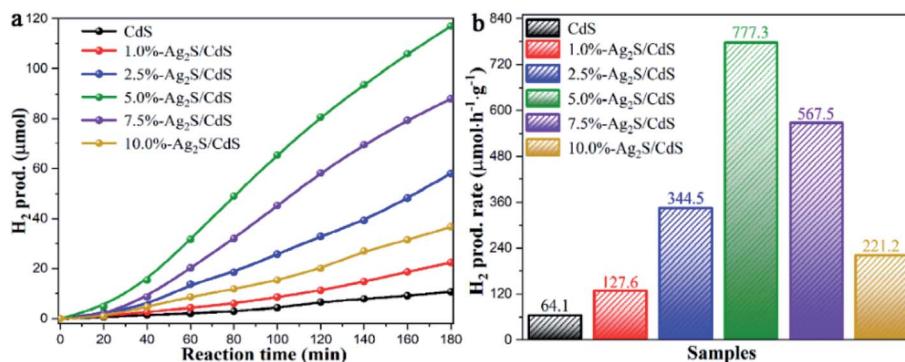


Fig. 1 (a) Time courses of photocatalytic  $H_2$  production; and (b) comparison of the  $H_2$  generation rate of samples over CdS and Ag<sub>2</sub>S/CdS nanocomposites under visible light irradiation.

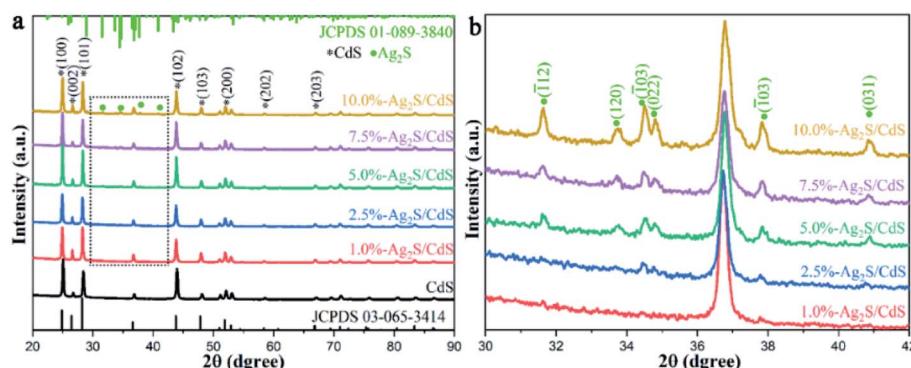


Fig. 2 (a and b) XRD patterns of CdS, and Ag<sub>2</sub>S/CdS nanocomposites.

the crystal structures of pure CdS and Ag<sub>2</sub>S/CdS nanocomposites. As shown in Fig. 2a, it is obvious that the sharp and strong characteristic diffraction peaks located at 24.9°, 26.5°, 28.2°, 36.6°, 47.8°, 51.0°, 58.4°, and 66.8° belong to the (100), (002), (101), (102), (103), (200), (202), and (203) planes of hexagonal CdS with high crystallinity (JCPDS no. 03-065-3414), respectively.<sup>18</sup> In addition to the diffraction peaks of pristine CdS, some new peaks at 31.5°, 33.7°, 34.4°, 34.7°, 37.7°, and 40.8° were also observed in the XRD patterns of (2.5%, 5.0%, 7.5%, 10%)-Ag<sub>2</sub>S/CdS nanocomposites (Fig. 2b). These peaks can be ascribed to the (112), (120), (121), (022), (103), and (031) faces of monoclinic Ag<sub>2</sub>S (JCPDS no. 01-089-3840), respectively.<sup>19</sup> This observation demonstrated that Ag<sub>2</sub>S has been successfully loaded onto CdS. However, the characteristic diffraction peaks of Ag<sub>2</sub>S are hardly identified in the XRD data of 1.0%-Ag<sub>2</sub>S/CdS nanocomposite which may due to the low content of Ag<sub>2</sub>S in the Ag<sub>2</sub>S/CdS nanocomposites. Therefore, it is necessary to further prove the existence of Ag<sub>2</sub>S in Ag<sub>2</sub>S/CdS nanocomposite by other characterization techniques.

To gain deep insight into the morphology and structure of CdS and Ag<sub>2</sub>S/CdS nanocomposites, the scanning electron microscopy (SEM) images were obtained. As shown in Fig. 3a, the morphology of a single CdS displays nanowire with approximately 5  $\mu\text{m}$  in length and has a clean and smooth surface. Compared with the smooth surface of CdS NWs, the

Ag<sub>2</sub>S/CdS nanocomposites remain uniform nanowires, except the covering with more and more nano-scaled nanoparticles on their surfaces, as the molar ratio of Ag<sub>2</sub>S to CdS increasing from 1.0% to 10.0% (Fig. 3a-f). Fig. 3(d<sub>1</sub>-d<sub>3</sub>) display the special distribution of Cd, S, and Ag elements for the sample of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite. The distribution of these elements is quite uniform and agrees with the shape of CdS NW, indicating that Ag<sub>2</sub>S NPs have well adhered onto the surface of CdS NWs. Simultaneously, the SEM results indicate the *in situ* ion exchange process of Ag<sup>+</sup> is an effective method to construct Ag<sub>2</sub>S-based cocatalyst.<sup>20</sup>

To further demonstrate the existence of Ag<sub>2</sub>S NPs on the surface of CdS NWs, transmission electron microscopy (TEM) was used to characterize the details of CdS NWs and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite. As shown in Fig. 4a, pure CdS NWs display a smooth surface with approximately 64 nm in diameter. This indicates that the aspect ratio of CdS NWs is quite high. This observation agrees well with the SEM measurements as mentioned above. And the large aspect ratio of CdS NW provides a high-speed channel for carrier transmission during photocatalytic process. Fig. 4b shows the high-resolution TEM (HRTEM) image of CdS NW. The interplanar distance between two adjacent crystal planes is measured to be 2.45  $\text{\AA}$  which is agreed well with the interlayer distance of (102) plane of CdS (Fig. 4b, b<sub>1</sub> and b<sub>2</sub>). Fig. 4c shows the low-magnified TEM image

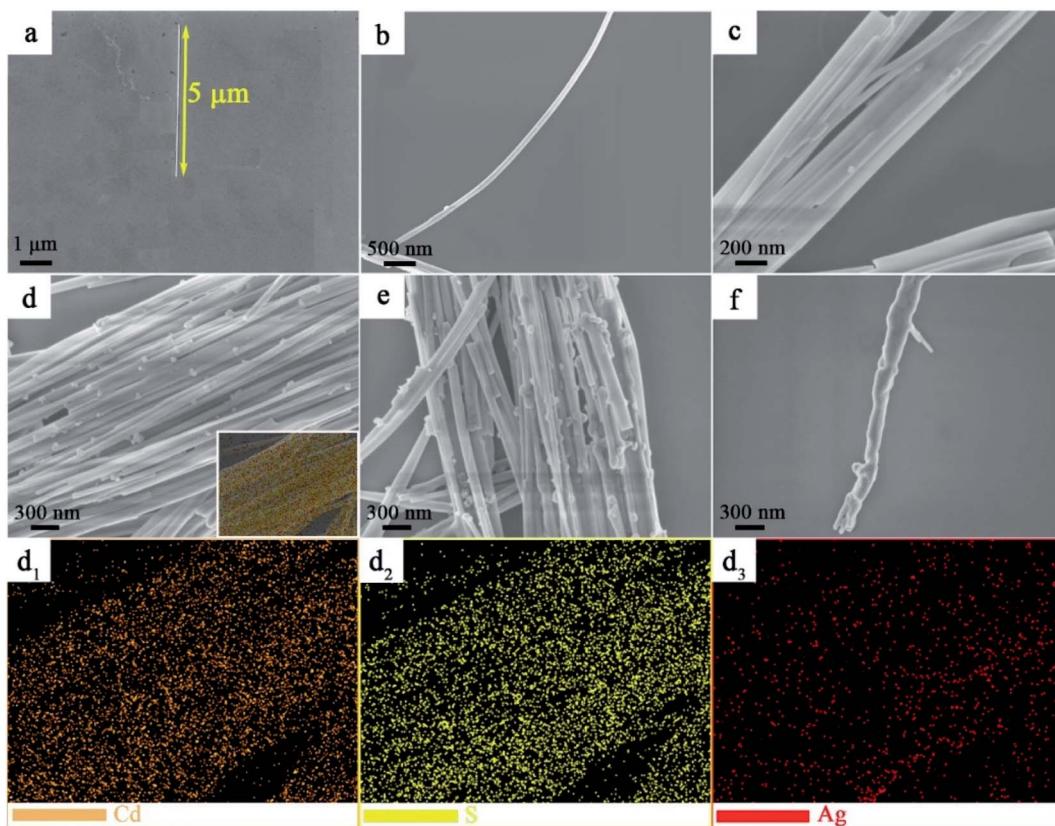


Fig. 3 SEM images of various samples: (a) CdS, (b) 1.0%-Ag<sub>2</sub>S/CdS, (c) 2.5%-Ag<sub>2</sub>S/CdS, (d) 5.0%-Ag<sub>2</sub>S/CdS, (e) 7.5%-Ag<sub>2</sub>S/CdS, (f) 10.0%-Ag<sub>2</sub>S/CdS, and (d<sub>1</sub>–d<sub>3</sub>) corresponding elemental mappings showing the ordered distribution of (d).

of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite for which the morphology is quite different to that of pure CdS NWs. It clearly shows that some irregular particles with sizes ranging from 20 to 80 nm are randomly located on the surface of CdS NWs for the 5.0%-Ag<sub>2</sub>S/CdS nanocomposite (Fig. 4c). Fig. 4d shows the HRTEM images of 5.0%-Ag<sub>2</sub>S/CdS, the *d*-spacing fringes of 2.45 Å and 2.05 Å are in good agreement with CdS (102) and Ag<sub>2</sub>S (103) plane, respectively. In addition, the consecutive lattice fringes between CdS NWs, and Ag<sub>2</sub>S NPs manifest the formation of desirable heterojunction structure at their interfaces.

Fig. 5 shows nitrogen adsorption–desorption isotherms and the corresponding pore-size distribution curves of pure CdS, 5.0%-Ag<sub>2</sub>S/CdS, and 10.0%-Ag<sub>2</sub>S/CdS nanocomposites. As shown in Fig. 5a, it can be seen that the nitrogen adsorption–desorption isotherms for these three samples are type IV with hysteresis H3 hysteresis loops at the *P/P*<sub>0</sub> range of 0.7–1.0 according to the IUPAC classification, indicating the existence of mesopores and large macropores.<sup>21</sup> The BET specific surface area, pore volumes, and average pore sizes of all of the samples are summarized in Table 1. The BET specific surface area (*S*<sub>BET</sub>) of the pure CdS nanowires is about 60.79 m<sup>2</sup> g<sup>-1</sup>. After 5.0% Ag<sub>2</sub>S is coated the *S*<sub>BET</sub> of 5.0%-Ag<sub>2</sub>S/CdS increased to 65.15 m<sup>2</sup> g<sup>-1</sup>. This is because the irregular Ag<sub>2</sub>S nanoparticles grow on the surface of the CdS NWs, resulting in expanded surface area. It was well-recognized that the large surface area contributes to absorbing more active species and reactants on its surface,

which is beneficial to effectively promoting the photocatalytic reaction.

As can be seen in the inset of Fig. 5b, the pristine CdS nanowires contain both mesopores (2–50 nm) and macropores (50–100 nm). The presence of small amount of Ag<sub>2</sub>S (5.0%) can introduce a lot of mesopores (2–50 nm) and thus decrease the pore volumes, and average pore sizes. However, for 10.0%-Ag<sub>2</sub>S/CdS, the further increase of the Ag<sub>2</sub>S content (from 5.0% to 10.0%) may lead to the adhesion and aggregation excessively of Ag<sub>2</sub>S on the CdS surface, which in turn reduces *S*<sub>BET</sub> and pore volumes. This analysis is in accordance with the SEM results.

Fig. 6a displayed the UV-vis diffuse reflectance spectrometer (DRS) of CdS NWs and Ag<sub>2</sub>S/CdS nanocomposites, which has been extensively used to evaluate the optical properties of semiconductor photocatalysts. It is worth noting that the attachment of Ag<sub>2</sub>S NPs on CdS NW usually causes the narrowing of the band gap. This effect results in the red shift of the cutting edge in UV-vis spectra compared with pure CdS NWs. According to Kubelka–Munk function *vs.* the band-gap energy, the absorption edge of pure CdS NWs and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite are located at 518 nm and 533 nm, which corresponds to a bandgap of *ca.* 2.39 eV and 2.33 eV. Besides, with the increment of Ag<sub>2</sub>S content, the absorption intensity of the Ag<sub>2</sub>S/CdS nanocomposites increase simultaneously in visible and near-infrared regions (520–1000 nm). The narrowed band-gap and increased absorption intensity of Ag<sub>2</sub>S/CdS



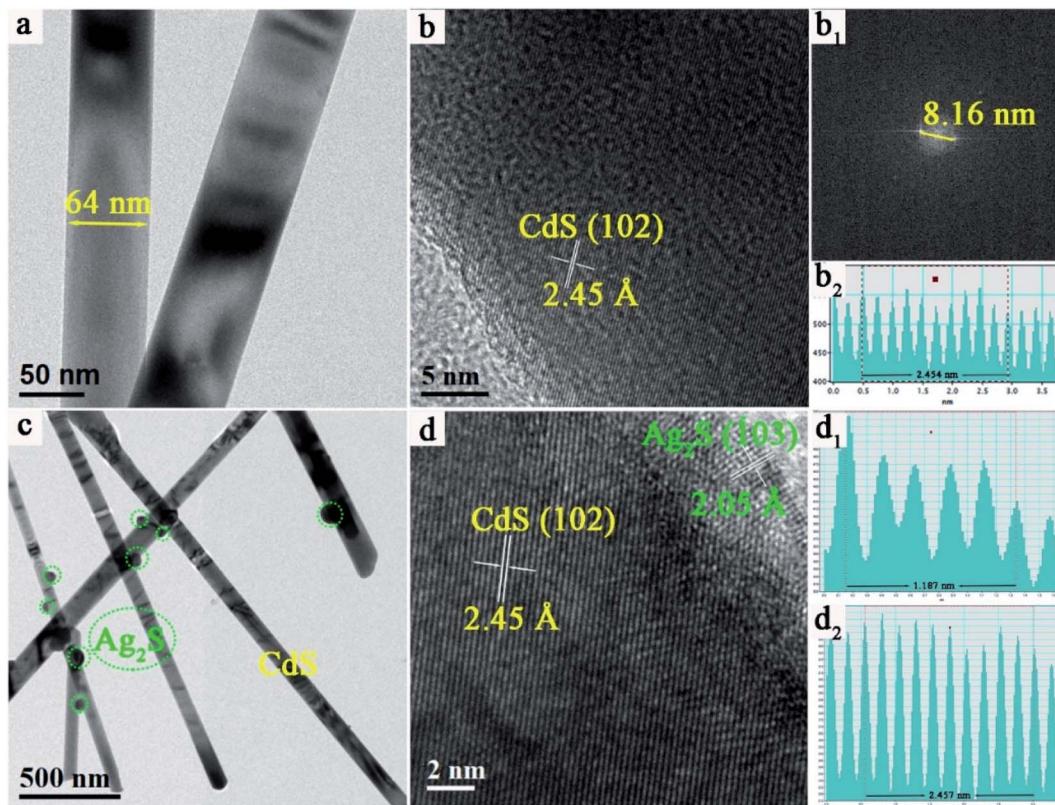


Fig. 4 (a and b) TEM/HRTEM results of CdS NWs, (b<sub>1</sub>) corresponding FFT patterns of (b) and (b<sub>2</sub>) profile plots of the calibration for measured the interplanar distance of CdS; (c and d) TEM/HRTEM images of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite and (d<sub>1</sub> and d<sub>2</sub>) profile plots of the calibration for measured the interplanar distance of CdS and Ag<sub>2</sub>S in (d).

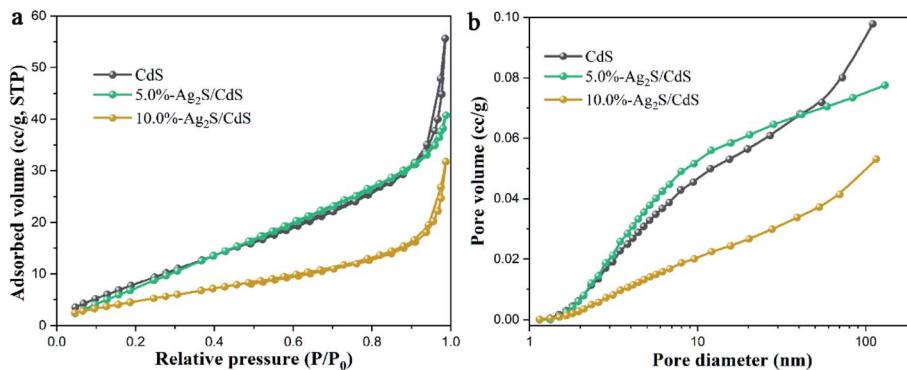


Fig. 5 (a) N<sub>2</sub> adsorption–desorption isotherms; and (b) the corresponding pore size distribution curves of CdS, 5.0%-Ag<sub>2</sub>S/CdS, and 10.0%-Ag<sub>2</sub>S/CdS.

nanocomposites can be ascribed to the formation of heterojunction and the high absorption cross section of Ag<sub>2</sub>S NPs (Fig. 6b).<sup>22,23</sup> These observations indicate that these nanocomposites can produce photoexcited electron–hole pairs under visible light irradiation, thus triggering consecutive chemical redox reactions.

To further investigate the surface chemical state and bonding configuration of samples, X-ray photoelectron spectroscopy (XPS) analysis was employed. As shown in Fig. 7a, both CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite contain Cd, S, O and C

elements. In addition, the characteristic peak of Ag was also detected, suggesting the existence of Ag<sub>2</sub>S in the Ag<sub>2</sub>S/CdS nanocomposite. Fig. 7b denoted the Cd 3d spectra of CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite, the two main peaks centered at 404.5 eV and 411.3 eV were thoroughly deconvoluted into Cd 3d<sub>5/2</sub> and Cd 3d<sub>3/2</sub> of Cd<sup>2+</sup>, respectively. In Fig. 7c, the binding energies at 161.0 eV and 162.2 eV for pure CdS, and at 161.6 eV and 162.2 eV for 5.0%-Ag<sub>2</sub>S/CdS nanocomposite are indexed to S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub> spin–orbit for S<sup>2–</sup>, respectively. The corresponding peaks of S 2p for 5.0%-Ag<sub>2</sub>S/CdS nanocomposite

**Table 1** The  $S_{\text{BET}}$ , pore volume, and average pore size of pure CdS nanowires and NiS/CdS nanocomposites

Sample	$S_{\text{BET}}$ ( $\text{m}^2 \text{ g}^{-1}$ )	Pore volume ( $\text{cm}^3 \text{ g}^{-1}$ )	Average pore size (nm)
CdS	60.79	0.10	2.32
5.0%-Ag <sub>2</sub> S/CdS	65.14	0.08	2.10
10.0%-Ag <sub>2</sub> S/CdS	27.35	0.05	2.11

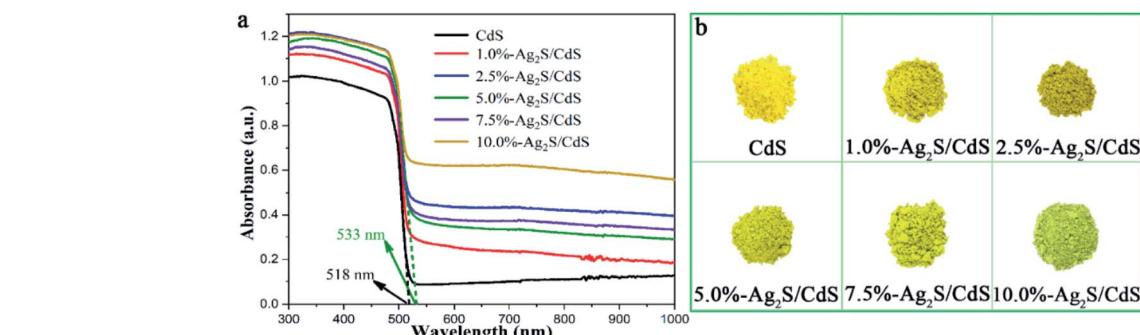


Fig. 6 (a) UV-vis DRS and (b) corresponding images of CdS and Ag<sub>2</sub>S/CdS nanocomposites.

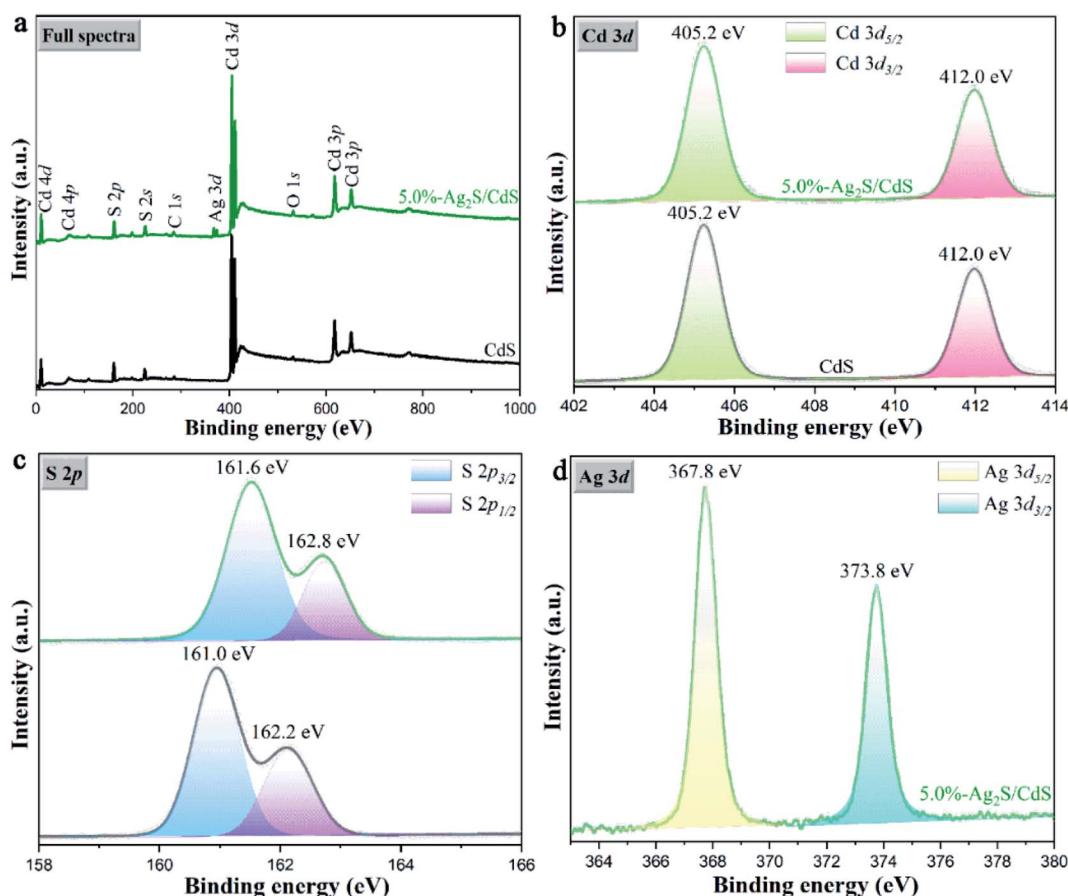


Fig. 7 (a) Full XPS survey spectra; and high-resolution XPS spectra of (b) Cd 3d, (c) S 2p and (d) Ag 3d of the CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite.

moved slightly toward higher binding energy than those of pure CdS. The shifting of peak positions may origin from the intimate interaction between Ag<sub>2</sub>S and CdS, which can result in the change in electron concentration on the surface of the photocatalysts. As given in Fig. 7d, two peaks located at 367.8 eV and 373.5 eV of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite could be ascribed to the Ag 3d<sub>5/2</sub> and Ag 3d<sub>3/2</sub> of Ag<sup>+</sup>, respectively.<sup>24</sup> The above XPS results further manifest the coexistence of CdS and Ag<sub>2</sub>S in 5.0%-Ag<sub>2</sub>S/CdS nanocomposite. Meanwhile, a heterojunction as

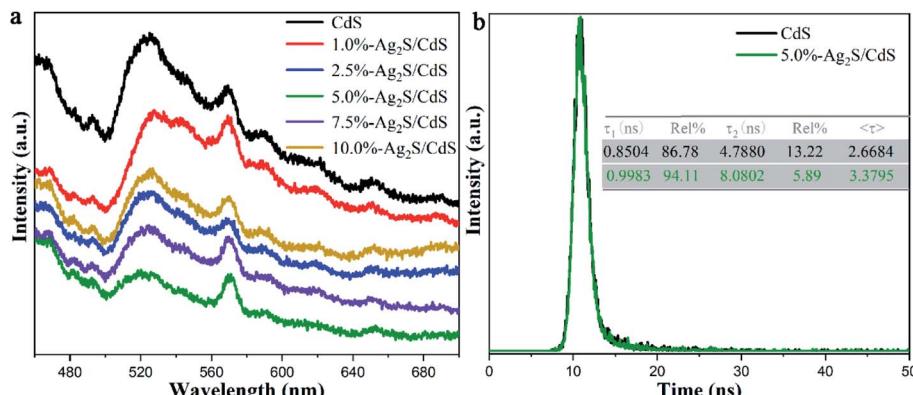


Fig. 8 (a) Steady-state PL spectra of CdS and Ag<sub>2</sub>S/CdS nanocomposites; and (b) TRPL decay of CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposites at an excitation wavelength of 525 nm.

being demonstrated by the HRTEM images shown in Fig. 5 is formed at the interface of the two components.

### 3.3. Photoelectric performance characterization

It is well-known that photoluminescence (PL) spectra can provide the information on separation and transfer process of photogenerated charge carriers. We carried out PL measurements to investigate the carrier behavior of CdS and Ag<sub>2</sub>S/CdS nanocomposites. The low PL intensity indicates a high electron–hole separation rate and excellent photocatalytic performance.<sup>25</sup> Fig. 8a is the steady-state PL spectra of all samples using an excitation wavelength of 380 nm, and two obvious emission peaks at about 525 and 570 nm appeared. The emission peak at 525 nm is assigned to the near-band-gap transition in CdS, and the other emission peak at 570 nm is due to the recombination of the electron–hole pairs near the surface of the photocatalysts.<sup>26</sup> It is worth noting that the peak intensity at 570 nm is strong. This phenomenon is ascribed to the long wire-like of CdS. Liu *et al.* discovered that the area of exposed nonpolar surface and degree of surface defects increase with an increase in aspect ratio.<sup>27</sup> In our work, the aspect ratio ( $\sim 78$ ) of long CdS nanowire is quite high, which means there is abundant surface defects on the surface of CdS. These surface defects can trap photo-induced electrons/holes, and thus decrease their probability of recombination. Meanwhile, compared with pure CdS, the PL spectra of Ag<sub>2</sub>S/CdS nanocomposites show similar but with weaker intensity, indicating their high-efficiency of photogenerated electron–hole separation. Among them, 5.0%-Ag<sub>2</sub>S/CdS nanocomposite shows the weakest PL spectral intensity, illustrating that the most efficient separation rate of photogenerated electrons and holes by the heterojunction effect between Ag<sub>2</sub>S and CdS.

Furthermore, the time-resolved photoluminescence (TRPL) character (Fig. 8b) of pure CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite were recorded to track the photo-physical behavior of charge carriers. The fluorescent decay profiles of both samples can be well fitted by two exponential terms, and their average lifetime was obtained according to the following equations:<sup>28,29</sup>

$$\langle \tau \rangle = \frac{B_1 \tau_1^2 + B_2 \tau_2^2}{B_1 \tau_1 + B_2 \tau_2} \quad (3)$$

where  $\langle \tau \rangle$ ,  $\tau_{i=1,2}$ , and  $B_{i=1,2}$  refer to the average lifetime, the decay time of the individual components, the corresponding amplitudes, respectively. Specially, the shorter lifetime,  $\tau_1$ , increased from 0.08504 ns for pure CdS to 0.9983 ns for 5.0%-Ag<sub>2</sub>S/CdS nanocomposite. In addition, the corresponding component increased from 86.78% to 94.11%. The longer lifetime,  $\tau_2$  of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite was increased to 8.0802 ns, but its component decreased from 13.22% to 5.89%. Using these parameters, an averaged lifetime of *ca.* 3.3795 ns for the sample of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite, which is remarkably longer than that of pure CdS NWs (2.6684 ns). The prolonged lifetime is closely related to the acceleration of electron transport, which can boost the probability of photoinduced electrons and holes participating in photo-redox process.

Electrochemical analysis such as transient photocurrent response, electrochemical impedance spectroscopy (EIS) Nyquist plots, linear sweep voltammetry (LSV), and Mott–Schottky plots were carried out to analyze the photogenerated carriers transfer properties of CdS and Ag<sub>2</sub>S/CdS nanocomposites. The transient photocurrent-time curves were measured by switching the irradiation light on–off for several runs. As displayed in Fig. 9a, a notable rise in the photoelectric response formed for all samples, indicating the high response of photoinduced electrons under visible-light stimulation. However, pure CdS shows a very weak photocurrent due to the fast electron–hole pairs recombination. And the photocurrent density becomes almost undetectable after several cycles of light stimulation due to the photocorrosion effect. In contrast, Ag<sub>2</sub>S/CdS samples show much higher photocurrent intensity and more reliable stability than that of pure CdS NW. We found that the sample of 5.0%-Ag<sub>2</sub>S/CdS nanocomposite shows the strongest photocurrent. This implies that the sample of 5.0%-Ag<sub>2</sub>S/CdS is most efficient for the photogenerated electron–hole pairs separation. This observation is consistent with PL spectra mentioned above.

To explore the influence of Ag<sub>2</sub>S on the electronic transport dynamics of CdS, EIS measurements are performed (Fig. 9b). The arc radius of Ag<sub>2</sub>S/CdS nanocomposites in the Nyquist plots are smaller than that of CdS, which indicates that the charge transfer resistance of Ag<sub>2</sub>S/CdS nanocomposites is lower than that of CdS, meaning the faster interfacial charge separation for



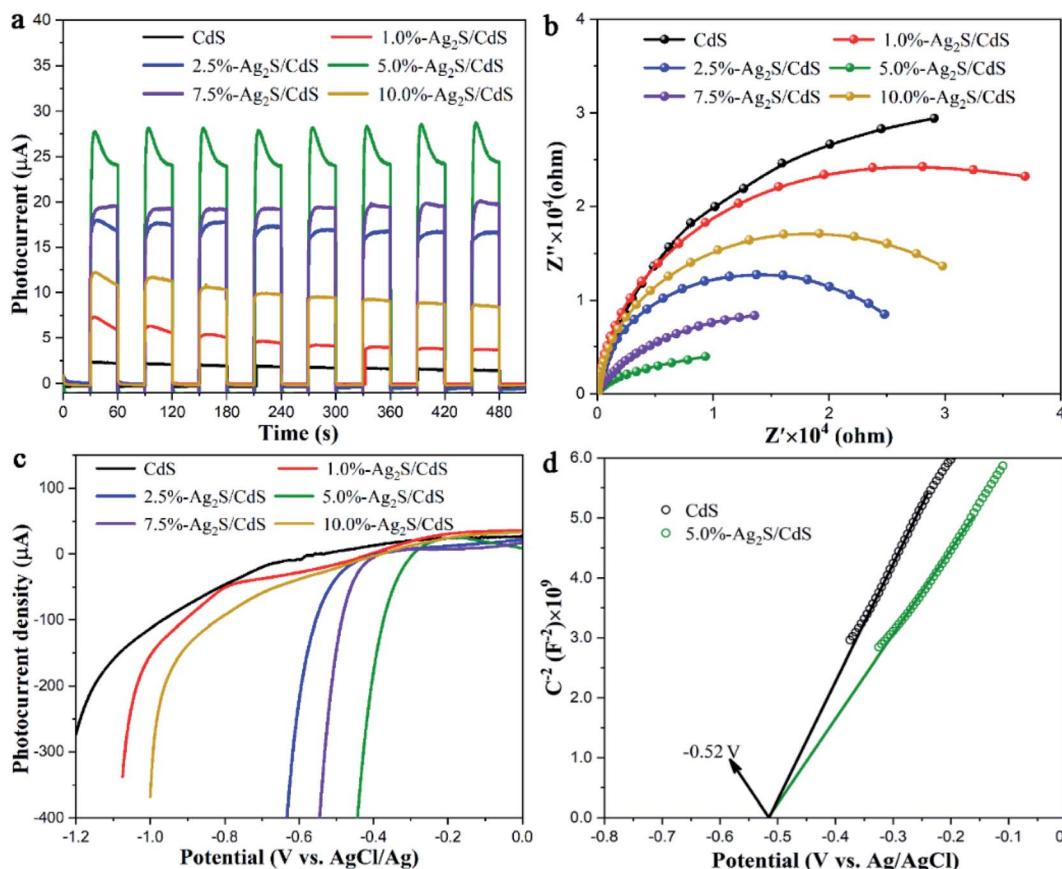


Fig. 9 (a) Transient photocurrent response; (b) EIS Nyquist plots; (c) LSV of CdS and Ag<sub>2</sub>S/CdS nanocomposites; (d) Mott–Schottky plots of CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposites.

the Ag<sub>2</sub>S/CdS nanocomposites. Concurrently, the 5.0%-Ag<sub>2</sub>S/CdS nanocomposite exhibited the smallest semicircle, implying that moderate Ag<sub>2</sub>S nanoparticles on CdS NWs can effectively enhance the charge separation efficiency.

To identify the key roles of Ag<sub>2</sub>S in improving the photocatalytic HER process, the polarization curves of CdS and Ag<sub>2</sub>S/CdS nanocomposites were also investigated. As shown in Fig. 9c, compared to the other samples, 5.0%-Ag<sub>2</sub>S/CdS nanocomposite had the lowest overpotential. Since the photocatalytic performance is closely related to overpotential,<sup>30</sup> the LSV results further support our conclusion that 5.0%-Ag<sub>2</sub>S/CdS nanocomposite is an excellent photocatalyst for HER. The results are in accordance with the photocatalytic activity and PL analysis.

Mott–Schottky experiment was applied to evaluate the carrier density and band positions of the CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite. As presented in Fig. 9d, the positive slope of  $C^{-2}$  vs.  $E$  plot indicates that both samples are n-type semiconductors.<sup>31</sup> And the carrier density can be calculated through taking the slope of the linear part in the Mott–Schottky plots into the following equation:

$$N_d = \frac{2}{\epsilon \epsilon_0 q} \left[ \frac{d \left( \frac{1}{C^2} \right)}{dV} \right]^{-1} \quad (4)$$

Here,  $N_d$  is the carrier density,  $\epsilon$  is the dielectric constant of Ag<sub>2</sub>S (7.77) and CdS (8.7),  $\epsilon_0$  ( $8.854 \times 10^{-12}$  F m<sup>-1</sup>) is the vacuum

permittivity,  $q$  ( $1.603 \times 10^{-19}$  C) is the electron charge, and  $V$  is the potential applied at the electrode, respectively. Using these parameters, the  $N_d$  were calculated as  $8.29 \times 10^{14}$  cm<sup>-3</sup> and  $1.15 \times 10^{15}$  cm<sup>-3</sup> for CdS and 5.0%-Ag<sub>2</sub>S/CdS nanocomposite, respectively. It's obvious that the electron density in the 5.0%-Ag<sub>2</sub>S/CdS nanocomposite much higher than that of CdS. This property can undoubtedly enhance H<sub>2</sub> evolution activity for Ag<sub>2</sub>S/CdS nanocomposites.

### 3.4. Photocatalytic mechanism

In order to understand the electron transfer between CdS and Ag<sub>2</sub>S in Ag<sub>2</sub>S/CdS nanocomposites, the work functions ( $W_F$ ) of CdS and Ag<sub>2</sub>S were obtained in terms of the UPS spectra. And the work function value was calculated in accordance with the following equation:

$$W_F = h\nu - E_{\text{cut-off}} \quad (5)$$

where  $h\nu = 21.22$  eV,  $E_{\text{cut-off}}$  is the cut-off edge of the secondary electrons in samples. As shown in Fig. 10a, the  $E_{\text{cut-off}}$  of CdS and Ag<sub>2</sub>S are 16.97 eV and 16.73 eV, respectively. According to the eqn (5), the calculated  $W_F$  of CdS and Ag<sub>2</sub>S are 4.25 eV and 4.48 eV, respectively. The corresponding schematic diagram of the energy band structures energy band structures about CdS and Ag<sub>2</sub>S are shown in Fig. 10b. Ag<sub>2</sub>S displays a relatively lower Fermi level than that of CdS. When CdS and Ag<sub>2</sub>S come into

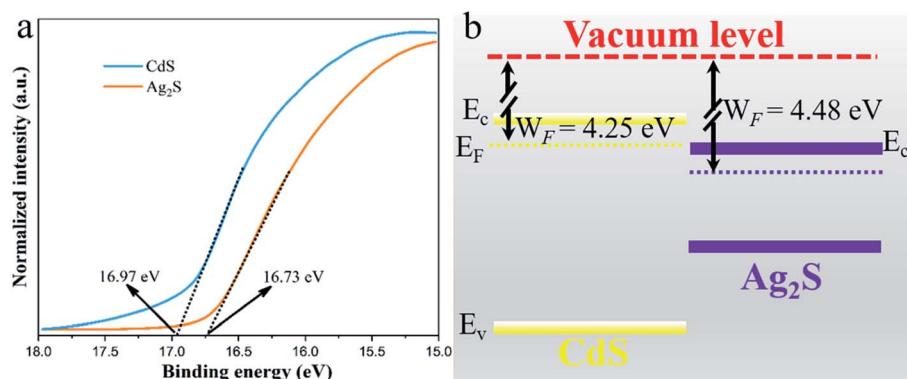


Fig. 10 (a) Ultraviolet photoelectron spectroscopy (UPS) spectra; and (b) the schematics of the energy band position of CdS and Ag<sub>2</sub>S.

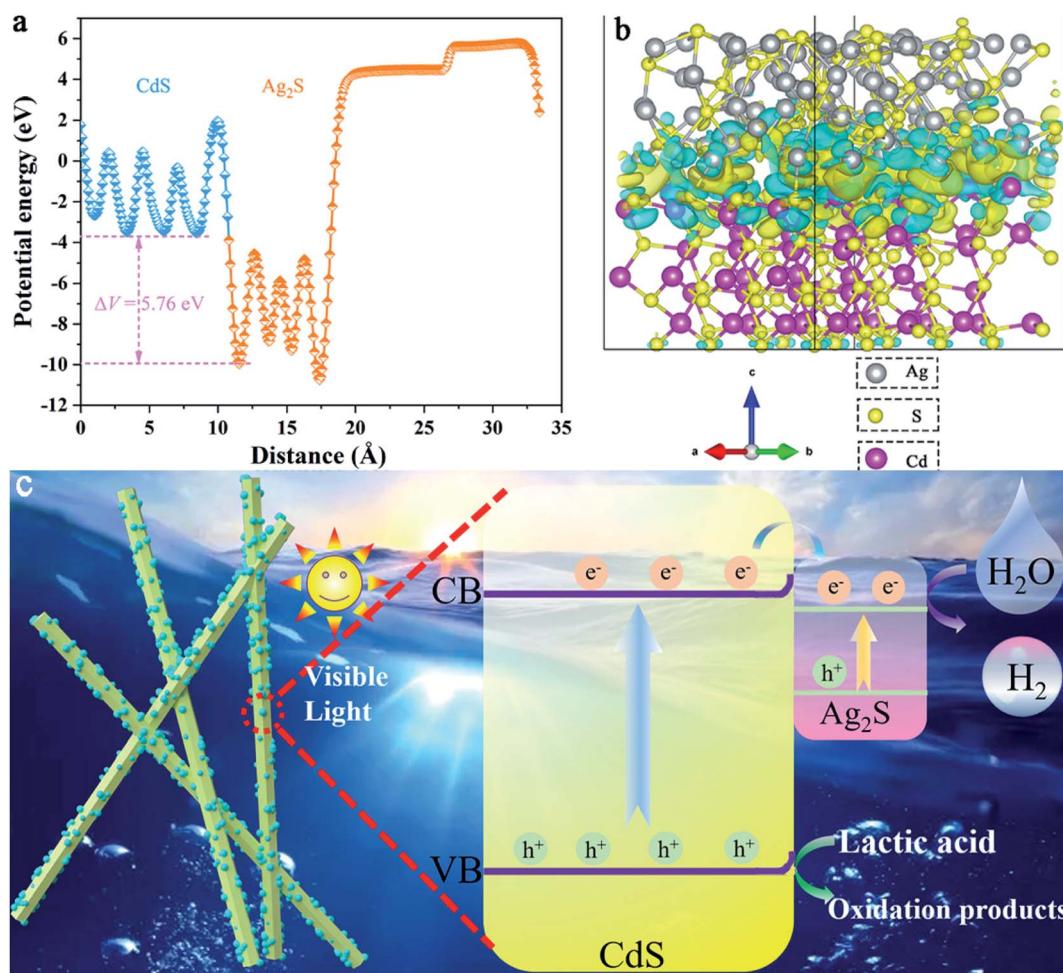


Fig. 11 (a) Potential drop across the interface; (b) the charge difference of Ag<sub>2</sub>S/CdS nanocomposites; (c) photocatalytic mechanism of H<sub>2</sub> generation with Ag<sub>2</sub>S/CdS composites.

contact, the electrons will diffuse from CdS to Ag<sub>2</sub>S to build up an equilibrium state between their Fermi levels.<sup>32,33</sup> The result indicates that there is a built-in electric field at the interface between CdS and Ag<sub>2</sub>S, and the conduction band position of Ag<sub>2</sub>S move upward to a more negative position. Thus, the photogenerated electrons in CdS tend to flow from CdS to Ag<sub>2</sub>S,

and then participate in the photocatalytic H<sub>2</sub> production from water splitting.

In addition, to better understand the charge transfer mechanism between the Ag<sub>2</sub>S/CdS nanocomposites, some theoretical calculations were carried out.<sup>34,35</sup> As shown in Fig. 11a, the potential drop ( $\Delta V$ ) across the Ag<sub>2</sub>S/CdS interface is 5.76 eV,

leading to the formation of a built-in electric field from the  $\text{Ag}_2\text{S}$  to the  $\text{CdS}$ . This built-in electric field is a key factor to enhances the separation of photogenerated electron–hole pairs. Fig. 11b gives the charge density difference of  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites. The cyan region and yellow region stand for charge depletion and accumulation near the heterojunction interface, respectively. It is found that both electrons and holes are located at the interface region of the  $\text{Ag}_2\text{S}/\text{CdS}$ , forming well-separated electron depletion and accumulation region, which is beneficial for the separation of electron–hole pairs.<sup>36,37</sup>

The flat-band potential ( $V_{fb}$ ) of pure  $\text{CdS}$  was obtained by extrapolating tangent to the  $X$ -intercept in the Mott–Schottky curve (Fig. 9b). It is worth mentioning that the introduction of  $\text{Ag}_2\text{S}$  does not alter the band structure of  $\text{CdS}$ . Both the  $V_{fb}$  of  $\text{CdS}$  and 5.0%– $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposite were about  $-0.63$  V vs.  $\text{Ag}/\text{AgCl}$ . Usually, the bottom of the conduction bands (CB) is more negative than the flat band potential for many n-type semiconductors. According to the transformation relation formula between  $\text{Ag}/\text{AgCl}$  electrode and normal hydrogen electrode (NHE):  $E_{\text{NHE}} = E_{\text{Ag}/\text{AgCl}} + 0.197$  and DRS results, the CB and valence band (VB) of  $\text{CdS}$  were  $-0.73$  V and  $1.66$  V vs. NHE, respectively. According to the previous reports, the CB and VB of  $\text{Ag}_2\text{S}$  are about  $-0.23$  and  $0.69$  V vs. NHE, respectively. Based on the above analysis, a photocatalytic mechanism of hydrogen generation with  $\text{Ag}_2\text{S}/\text{CdS}$  composites is proposed (Fig. 11c). Under visible light irradiation, both  $\text{CdS}$  and  $\text{Ag}_2\text{S}$  samples are simultaneously excited to produce photogenerated electrons and holes. Meanwhile, the CB of  $\text{CdS}$  is more negative than that of  $\text{Ag}_2\text{S}$ , the photogenerated electrons in the CB of  $\text{CdS}$  can be rapidly migrated to the conduction band of  $\text{Ag}_2\text{S}$ . Thus, these can participate in HER through the abundant active sites in  $\text{Ag}_2\text{S}$ . These results confirm that the fast charge separation and transportation in the hetero-structural nanomaterials can greatly boost photocatalytic  $\text{H}_2$  production.

## 4. Conclusion

In summary,  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposites with high hydrogen evolution ability are successfully synthesized through a hydrothermal and subsequent *in situ* ion exchange method. The best photocatalytic activity of  $\text{H}_2$  production of  $777.3 \mu\text{mol h}^{-1} \text{g}^{-1}$  in lactic acid solution was achieved for  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposite with  $\text{Ag}_2\text{S}$  content ratio of 5.0%. The preeminent photocatalytic activity is mainly attributed to the driving force provided by the potential decline at the heterojunction interface, which can promote the photoinduced electrons directionally transfer from  $\text{CdS}$  NWs to  $\text{Ag}_2\text{S}$  NPs to participate in the HER. Meanwhile, benefitting from the narrow band gap of  $\text{Ag}_2\text{S}$ , the light absorption of the samples extends from visible light to infrared light, meaning their efficient light utilization efficiency. Furthermore, the photocatalytic mechanism of the  $\text{Ag}_2\text{S}/\text{CdS}$  nanocomposite is investigated in detail. Thus, this work sheds light on the synthesis of  $\text{CdS}$ -based nanocomposites for environmental and energetic applications.

## Author contributions

Congrong Lu: conceptualization, data curation, investigation, formal analysis. Shiwen Du: methodology, data curation, writing-original draft. Yanfei Zhao: supervision, data curation, software. Qi Wang: conceptualization, investigation, resources, methodology. Kuankuan Ren: supervision, visualization, software. Chunhe Li: conceptualization, formal analysis, funding acquisition, writing-review & editing. Weidong Dou: validation, resources, funding acquisition, project administration.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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