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In situ hydrothermal synthesis of nickel cobalt sulfide nanoparticles embedded on nitrogen and sulfur dual doped graphene for a high performance supercapacitor electrode†

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Nickel cobalt sulfide nanoparticles (NCS) embedded onto a nitrogen and sulfur dual doped graphene (NS-G) surface are successfully synthesized *via* a two-step facile hydrothermal process. The electrical double-layer capacitor of NS-G acts as a supporting host for the growth of pseudocapacitance NCS nanoparticles, thus enhancing the synergistic electrochemical performance. The specific capacitance values of 1420.2 F g⁻¹ at 10 mV s⁻¹ and 630.6 F g⁻¹ at 1 A g⁻¹ are achieved with an impressive capability rate of 76.6% preservation at 10 A g⁻¹. Furthermore, the integrating NiCo₂S₄ nanoparticles embedding onto the NS-G surface also present a surprising improvement in the cycle performance, maintaining 110% retention after 10 000 cycles. Owing to the unique morphology an impressive energy density of 19.35 W h kg⁻¹ at a power density of 235.0 W kg⁻¹ suggests its potential application in high-performance supercapacitors.

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1. Introduction

Over the past few decades, electrical energy demands have been increased by the advancement of portable electronic devices and luxurious “standards of living” technologies. The requirements of high-performance energy storage/supply have been set for next-generation electronic devices, electronic vehicles and electronic household appliances such as high-power-energy output, long-life cycles, safety, compact size, and light weight.¹ Supercapacitors possess promising characteristics as an electrical energy storage/supply including high power energy output, fast charge features, low cost and high stability with a long lifetime.²

For charge storage mechanisms, supercapacitors (SCs) can be divided into two types of storage mechanism, namely, pseudocapacitors (PCs) and electrical double layer capacitors (EDLCs).³ PC is identified as the reversible faradaic reaction between surface materials and electrolyte, meanwhile EDLC stores the electrical energy by electrostatic absorption between

ion/charge diffusion from electrolyte and the surface materials.^{4,5} Recently, extensive materials have been explored as high-performance materials for PC such as conductive polymers (polyaniline,^{6,7} polypyrrole^{8,9} and poly(2,3-dihydrothieno-1,4-dioxin) or PEDOT¹⁰) and transition metals (ZnO,¹¹⁻¹⁴ Ni(OH)₂,¹⁵ V₂O₅,¹⁶ TiO₂,¹⁷ MnO₂,¹⁸ NiS₂ and CoS₂ (ref. 19)). In particular, nickel and cobalt sulfide have been chosen as the electrode material because of their rich electroactive sites, multiple redox reaction, and high theoretical capacitances.¹⁹⁻²¹ Many attempts have been made to develop the high pseudocapacitance materials with nanostructure. The laser irradiation technique was employed to synthesize nickel sulfide nanostructures, demonstrating the ultra-high specific capacitance of 3761 F g⁻¹ at 10 mV s⁻¹.²² A recent development on synthesis of zinc-cobalt sulfide nanoparticles *via* the hydrothermal process using hybrid zeolitic imidazolate framework (ZIF) as template reveal the ultra-high capacitance of 1646 F g⁻¹ at 1 A g⁻¹ with capacitance retention of 89% after 2000 cycles.²³ Mesoporous carbons with parallel channel structure decorated with nickel cobalt sulfide (NiCoS) nanoflower were successfully synthesized by Al leaching following by hydrothermal process. This ordered mesoporous carbon built conductive network with NiCoS, yielding the surprising electrochemical performance with the specific supercapacitance of 1757 F g⁻¹ at 1 A g⁻¹.²⁴

Nevertheless, even though there are many reports mentioned an excellent performance of metal sulfide pseudocapacitors in term of the high electrical conductivity and high capacitance values, but their structure stability of nickel-cobalt sulfide have been limited in ion diffusion, resulting in lack of reversible redox reaction at high current densities.^{16,25} In order to solve

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this issue, carbonaceous materials have been introduced as host materials, expecting to increase active sites and electrochemical stability.²⁶ Graphene-based electrode has been recently attended for EDLC and hybrid supercapacitors.²⁷ Interestingly, the doping process of graphene with heteroatoms (such as nitrogen, sulfur, phosphorous, *etc.*) was reported as an effective method to enhance its electrochemical storage performance, arising from redox surface functionality of heteroatom.²⁸ Its great potential due to not only possessing a high electrical surface area with redox reaction but also the ability to functionalize on the surface to improve the interaction with electrolyte. Sulfur doped graphene was eco-friendly synthesized using sodium thiosulphate as sulfur source and exfoliation catalyst, presenting the high specific capacitance of 320 F g⁻¹ at 3 A g⁻¹ with cycling efficiency of 92% after 1500 cycles.²⁹ Sulfur and nitrogen dual doped graphene using L-cysteine as doping agent was also reported as an excellent reversibility SCs, showing a cycling efficiency of 83.54% after 5000 cycles.³⁰

Therefore, to develop nanostructure of nickel cobalt sulfide (NCS) onto nitrogen, sulfur dual doped graphene (NS-G) surface, the designed facile two-step process of *in situ* hydrothermal process with low-cost effectiveness is proposed. The morphology-crystal-structure of an intermediate nickel/cobalt carbonate hydroxides mixture is developed in presence of thiourea and graphene oxide through mild hydrothermal process. The influence of coexisting of nickel cobalt sulfide and nitrogen, sulfur dual doped graphene surface is investigated in term of electrochemical performance, capacitive rate performance and cycle stability. Thus, NCS/NS-G composites is examined as potentially an ideal candidate utilizing for hybrid supercapacitors.

2. Experimental

2.1 Materials

Nickel(II) nitrate hexahydrate (Ni(NO₃)₂·6H₂O, >98%), hexamethylenetetramine (HMTA, >99%), sodium dodecyl sulfate (SDS, >99%) and graphite powder (particle size ≤20 μm) were supplied from Sigma-Aldrich. Cobalt(II) nitrate hexahydrate (Co(NO₃)₂·6H₂O, >99%) was purchased from Ajax Finechem. Vulcan XC-72R carbon black (CB, average particle size of 50 nm) and PD-70T carbon fiber paper (CFP) were obtained from Fuel Cell Store. Other chemical reagents used were of analytical grade.

2.2 Preparation of graphene oxide aqueous dispersion

Graphite oxide was synthesized by our modified Hummers' method. Briefly, 10 g of graphite powder and 5 g of sodium nitrate were mixed in 300 mL of concentrated sulfuric acid at 10 °C. Subsequently, 30 g of potassium permanganate were slowly added into the mixture and continually stirred at 10 °C for 4 h. After that, the reaction mixture was slowly poured into 350 ml of mixed ice water with the addition of H₂O₂. The filtrated products were flushed and neutralized with DI water before drying at 60 °C for 24 h. GO aqueous dispersion was prepared in DI water (2 mg ml⁻¹) with an ultrasonic bath (50 Hz) for 3.5 h.

2.3 Preparation of nickel cobalt carbonate hydroxide (NiCoCH)

10 ml of 0.5 M Ni(NO₃)₂·6H₂O and 20 ml of 0.5 M Co(NO₃)₂·6H₂O were mixed by control the mixture solution volume of 30 ml. 1.31 g of HMTA and 0.52 g of SDS were added and for 30 minutes. The mixture was transferred into 100 ml Teflon line autoclave and added more 20 ml of DI water. Then, the reactor was conducted at 140 °C for 14 hours. The obtained solid was filtrated and washed several times with DI water and ethanol. Finally, the obtained sample was dried at 60 °C for 24 hours.

2.4 Preparation of nickel cobalt sulfide/nitrogen and sulfur dual doped graphene (NS-G/NCS) composites

0.02 g of NiCoCH was dispersed in 20 ml of DI water. Then, 30 ml of 2 mg ml⁻¹ graphene oxide aqueous dispersion and 0.6 g thiourea was added. The mixture was sonicated for 30 minutes and transferred into a 100 mL Teflon line autoclave. The hydrothermal reaction was employed at 140 °C for 24 h. Finally, the obtained product was washed by DI water and dried using the freeze dry technique.

2.5 Characterization techniques

A scanning electron microscope equipped with energy dispersive X-ray spectrometer (SEM-EDS, JEOL, JSM-IT500HR, Japan) was used to observe the sample morphology and analyze the surface element of samples. The chemical valence states of the samples were characterized by X-ray photoelectron spectroscopy (XPS, Bara Scientific, AXIS ultra DLD, UK). X-ray diffraction analysis (XRD, BRUKER, D8 ADVANCE, USA) was performed by Cu-Kα radiation (1.5406 Å) at 0.02° per minutes ranging from 5° to 80°.

2.6 Preparation of working electrode samples

Working electrode samples were prepared by mixing the composite samples of active materials, carbon black, and PTFE at 70 : 20 : 10 by weight, respectively. The solid mixtures were mixed with ethanol to obtained slurry mixture and coated on carbon fiber paper sized at 1 × 1 cm². The electrode samples were obtained after drying at 60 °C for 24 hours.

2.7 Electrochemical characterization

Cyclic voltammetry (CV), galvanostatic charge–discharge (GCD) and electrochemical impedance spectroscopy (EIS) techniques were used to investigate the electrochemical properties of samples. Potentiostat/galvanostat instrument (Metrohm Autolab, PGSTAT204, Netherlands) configuring with three electrode systems including of sample electrode as working electrode, platinum plate as counter electrode and Ag/AgCl as reference electrode was performed in 6 M of KOH electrolyte. CV was investigated in potential range 0 V to 0.55 V, -0.3 V to 0.45 V and -0.1 V to 0.45 V for NS-G, NiCoCH and NS-G/NCS electrode, respectively, with difference scan rate from 10 to 100 mV s⁻¹. GCD was measured in potential range 0 V to 0.44 V, -0.2 V to 0.37 V and -0.1 V to 0.4 V for NS-G, NiCoCH and NS-G/NCS electrode, respectively, with current density ranging from 1 to



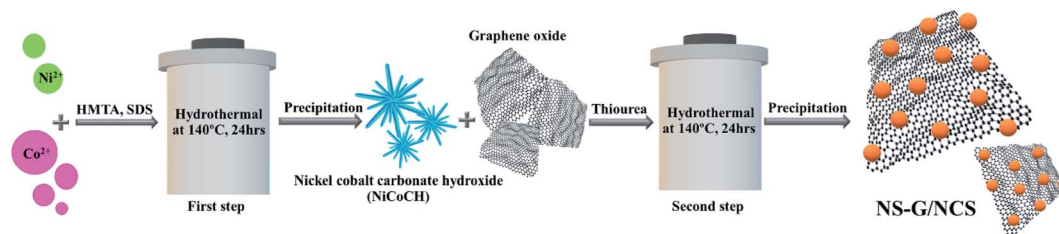


Fig. 1 Schematic of facile two-step hydrothermal synthesis of NS-G/NCS.

10 A g^{-1} . EIS was monitored at 5 mV over a frequency range from 0.01 Hz to 100 kHz in 1 M of KOH solution. The cycle stability was investigated by battery tester (Neware, CT4000). The symmetrical cell with CR2032 battery cell is fabricated for the GCD cycle stability, employing at 0.1 A g^{-1} from 1 to 10 000 cycles in 6 M KOH electrolyte. The specific capacitance (C_{sp}) values were calculated from GCD result as following the eqn (1).^{31,32} The energy density (E) and the power density (P) were calculated from eqn (2) and (3).^{33,34}

$$C_{\text{sp}} = \frac{I(t)}{m(\Delta V)} \quad (1)$$

$$E = \frac{1}{2} C_{\text{sp}} (\Delta V)^2 \quad (2)$$

$$P = E/T \quad (3)$$

where I is constant discharge current (A), t is the discharging time (s), m is mass of active material sample (g), ΔV is window potential (V) and T is discharging time (h).

3. Results and discussion

3.1 Morphological characterization

Fig. 1 schematically describes a designed facile two-step hydrothermal synthesis for nickel cobalt sulfide nanostructure embedding on nitrogen and sulfur dual doped graphene surface. Firstly, a mixture of nickel/cobalt carbonate hydroxides precursor is synthesized *via* a straightforward hydrothermal approach in the presence of Ni^{2+} , Co^{2+} , SDS and HMTA aqueous solution. Subsequently, the phase-structure-morphology of a mixture of nickel/cobalt carbonate hydroxides nanoparticle is developed to embed onto NS-G surface *via* second step of hydrothermal process in presence of graphene oxide aqueous dispersion and thiourea. Fig. 2 illustrates the representative surface morphologies of NS-G, nickel/cobalt carbonate hydroxide (NiCoCH) and NS-G/NCS. As seen in Fig. 2(a), NS-G manifestly shows the exfoliated layers of individual graphene sheets in 3D network structure, benefiting for electro-absorption of ion/charge electrolytes on surface, which is an excellent feature for of EDLC type materials.^{8,35,36} Fig. 2(b) displays the sea-urchin-like structure of NiCoCH, which is synthesized *via* the first step of hydrothermal process.

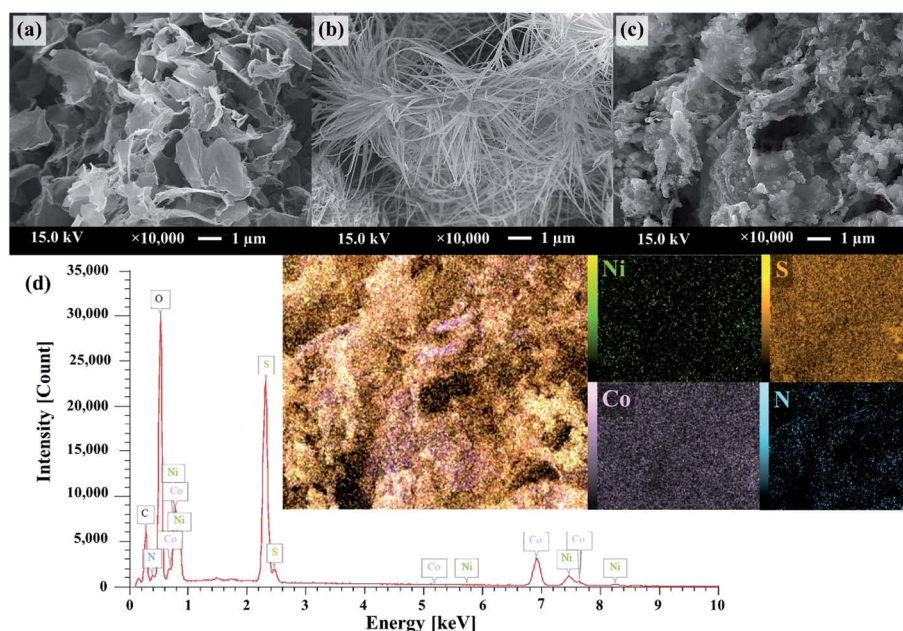


Fig. 2 SEM micrograph of NS-G (a); NiCoCH (b); NS-G/NCS (c) and EDS-mapping image of NS-G/NCS (d).



Table 1 Elemental percentage of NS-G/NCS

Elements	Mass (%)	Atom (%)
Nickel	7.81 ± 0.06	2.69 ± 0.02
Cobalt	18.5 ± 0.08	6.33 ± 0.03
Sulfur	17.9 ± 0.03	11.3 ± 0.02
Nitrogen	1.52 ± 0.02	2.19 ± 0.03

Interestingly, the crystallinity-morphology-structure of the urchin-like structure NiCoCH precursor can be developed *via* the second step of hydrothermal process in presence of graphene oxide and thiourea. Fig. 2(c) presents the embedded nickel-cobalt nanoparticle on graphene surface. Obviously, the second step approach of hydrothermal process can create the well uniform of highly dense Ni-Co nanoparticles embedded on NS-G. Furthermore, the element compositions of the sample composites were analyzed using energy dispersive X-ray spectrometer. The element distribution maps of NCS/NS-G sample is presented in Fig. 2(d) revealing the distribution of Ni, Co, S and C elements, confirming the compositions of Ni-Co complex on graphene surface. The atom percentage of element composition are also exhibited in Table 1.

3.2 Structural characterization

X-ray diffraction is used to investigate the crystal phase of the as-prepared samples. Fig. 3(a) exhibits the two obvious diffraction peaks of NS-G at 2θ of 23.7° and 42.6°, attributing to (002) and (001) reflection of graphitic carbon phase.^{29,37} The XRD pattern of as-prepared NiCoCH precursor in Fig. 3(b) reveals the diffraction peak at 2θ of 9.9°, 17.2°, 26.6°, 33.7° and 59.5°, which are a good indexed with cobalt carbonate hydroxide (PDF 00-048-0083) and nickel carbonate hydroxide (JCPDS no. 35-0501).³⁸ Fig. 3(c) reveals the diffraction peaks of nickel cobalt sulfide at 2θ of 26.8°, 32.1°, 38.2°, 50.4° and 55.2°, corresponding with PDF 00-020-0782 reference.^{20,39} Nevertheless, the presence of the signal diffraction pattern of the mixed nickel(II) sulfate hexahydrate and cobalt(II) sulfate hydrate compositions are also observed, implying the coexisting of impurities phase in nickel cobalt sulfide.

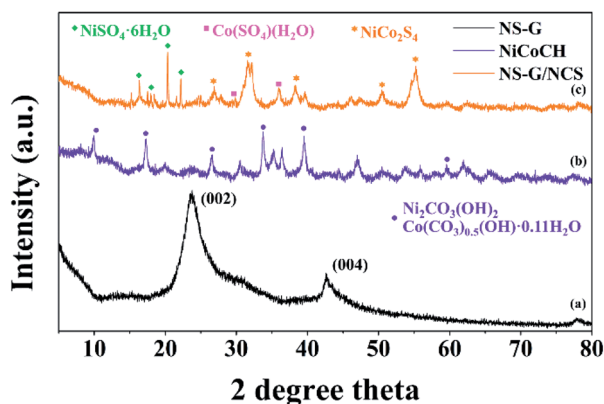


Fig. 3 XRD patterns of NS-G (a); NiCoCH (b) and NS-G/NCS (c).

The valence states and chemical compositions of NCS/NS-G sample were investigated by XPS technique. Fig. 4(a) reveals the survey spectra of NCS/NS-G, ascribing the existing of C, N, S, Ni, Co and O atoms. The deconvoluted C 1s spectrum identifies the presence graphene with nitrogen and sulfur group structures (Fig. 4(b)). The four peaks deconvolution of C 1s spectrum are observed at 285.0, 285.7, 286.5 and 287.8 eV, corresponding with sp^2 hybridized C=C bond, sp^3 hybridized C-O/C-S/C-N, -C-O and -C=O bonding, respectively.^{30,37} In addition, the calculated C/O ratio of NCS/NS-G sample is 18, implying the very low content of oxygen functionality in NCS structure and confirming the effective hydrothermal reduction process in presence of thiourea.⁴⁰ N 1s spectrum in Fig. 4(c) can be well fitted to the typical nitrogen doped graphene, which is deconvoluted into three individual peaks consisting of pyridinic N (399.2 eV), pyrrolic N (400.6 eV) and quaternary N (402.0 eV), respectively.^{30,37} Fig. 4(d) reveals the obvious two major peaks of S 2p spectrum at binding energies of 163.4 and 164.3 eV, which are corresponded with the sulfur binding in C-S bond and conjugated -C=S- bond in aromatic carbon, respectively.^{30,37,41} The other two obvious peaks at around 170.1 eV and 169.0 eV can be assigned to -C-SO_x-C- bonding, which are mainly derived from oxysulfide species.^{42,43} It is evidently seen that all mentioned XPS spectra results above suggest the successful doping of nitrogen and sulfur heteroatom onto graphene structure. For Ni 2p spectra, Fig. 4(e) illustrates two spin-orbitals at Ni 2p_{1/2} and Ni 2p_{3/2} with satellite peaks (noted as "Sat.") at binding energy of 874.6 and 857.1 eV, respectively, ascribing to Ni²⁺ characteristic.^{40,44,45} For Co 2p spectrum, the peaks at 798.5 eV and 794.1 eV are identified with the valence states of +2 and +3 of Co 2p_{1/2}, respectively. While, the peaks at 782.3 and 779.2 eV are respectively assigned to valence states of +2 and +3, respectively (Fig. 4(f)).⁴¹ The peaks 803.5 eV and 786.8 eV are ascribed to the satellite peaks of Co 2p_{3/2} and Co 2p_{1/2}, respectively. In addition, the peak of S 2p spectrum at 161.8 and 162.3 is ascribed to 2p_{3/2} and 2p_{1/2} electrons in metal sulfide, respectively, confirming that the S element is in the divalent state (Fig. 4(d)). These results suggest the presence of the mixed valence state of Ni²⁺, Co²⁺ and Co³⁺ in the obtained product of NCS/NS-G composites.

3.3 Electrochemical measurements

Cyclic voltammograms (CV) study has been used to evaluate the influence of NS-G, NiCoCH and NS-G/NCS electrode samples on electrochemical performance and to evaluate the specific capacitance of the as-prepared sample electrodes. Fig. 5(a) demonstrates cyclic voltammograms of NS-G, NiCoCH and NS-G/NCS. CV curve of mixed NiCoCH electrode exhibits the one anodic peak at 0.35 V and one cathodic peak at -0.15 V, indicating that the capacitive response corresponds with M-O-OH and M carbonate hydroxide (M represents Ni or Co) in sample electrodes.^{46,47} This pair redox peak of reversible faradaic redox reaction suggests the faradaic reaction in pseudo capacitance behavior of storage charge mechanism.⁴⁴ NS-G shows a presence of a small slope oxidation peak at 0.29 and 0.42 V and reduction peaks at 0.19 and 0.24 V, arising from the heteroatom effect of nitrogen and sulfur doping on graphitic carbon of



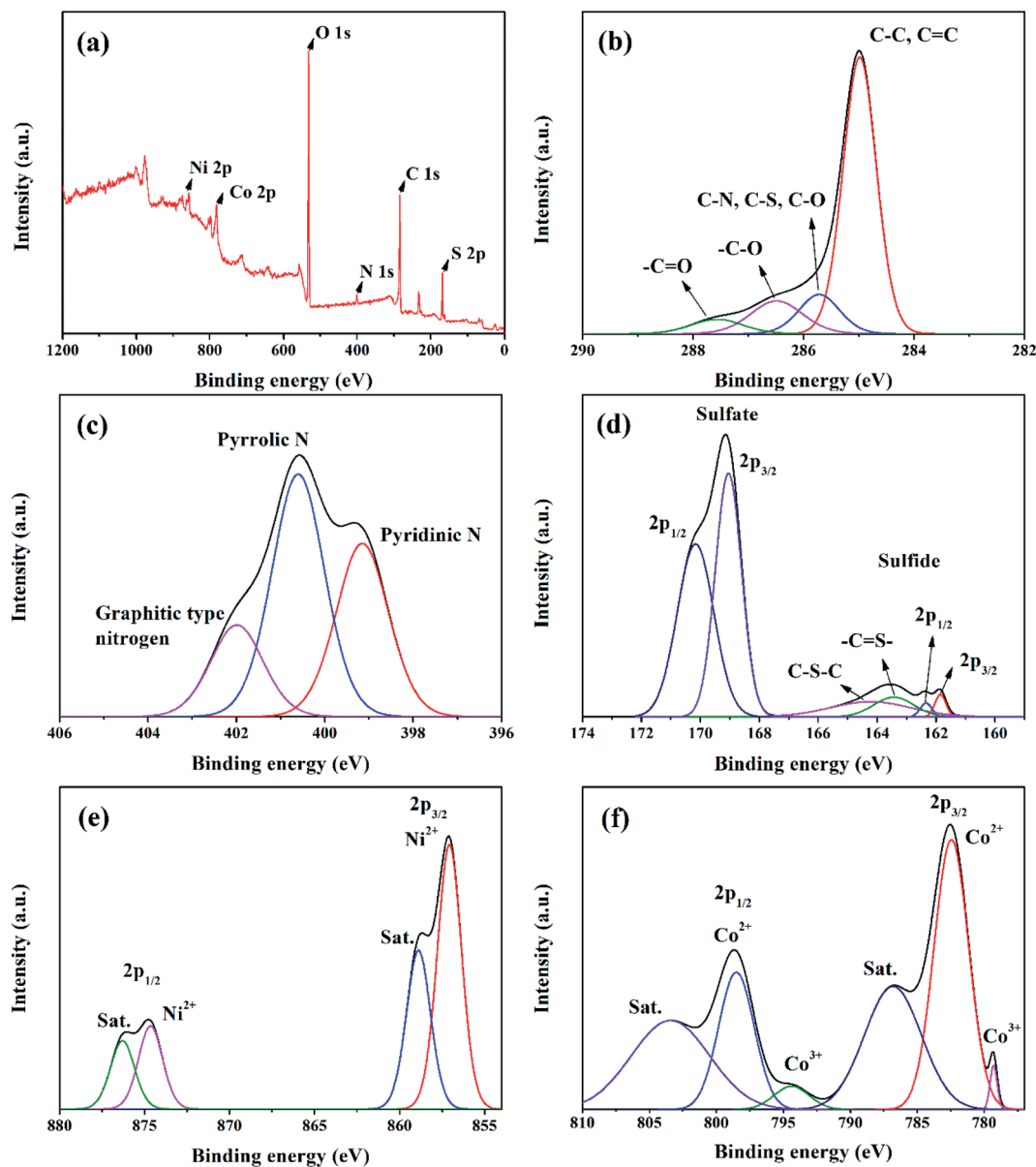
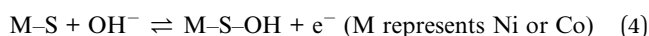


Fig. 4 XPS spectra of NS-G/NCS composites; survey spectrum (a), C 1s spectrum (b), N 1s spectrum (c), S 2p spectrum (d), Ni 2p spectrum (e), and Co 2p spectrum (f).

graphene structure.^{15,29} Meanwhile, NCS/NS-G electrode sample reveals a pair of redox peaks, attributing to faradaic reactions of M-S/M-S-OH pairs (M = Ni, Co). The two obvious redox regions are observed at between 0.1 to 0.3 V for oxidation and between 0.04 to 0.14 V for reduction, resulting from the reversible oxidation-reduction of metal-sulfur-hydroxide in KOH during given voltage potential (seen eqn (4)).^{19,48}



To verify the specific capacitance values, the area in CV curve is typically analyzed to evaluate the electrochemical active surface area (ECSA), indicating a sufficient number of electro-active sites during redox reactions and an electro-absorption

efficiency between active surface materials and ion/charge from electrolyte.^{49,50} Obviously, the introduce NCS embedded onto NS-G surface reveal the influence on the amount of charge storage as increasing in CV area characteristic due to the synergistic results of both EDLC and PC. The distinction in the current densities response for each sample electrode can be easily observed, revealing that NCS/NS-G electrode occupies the highest ECSA relating to the other electrode samples. The specific capacitance values of NS-G, NiCoCH and NCS/NS-G electrode samples are 467.1, 549.5 and 1420.2 F g^{-1} at 10 mV s^{-1} , respectively. Fig. 5(b) shows the calculated specific capacitance by CV of the as-prepared electrode samples as a function with scan rate ranging from 10 to 100 mV s^{-1} . Obviously, all sample electrodes reveal the decrease in specific capacitance



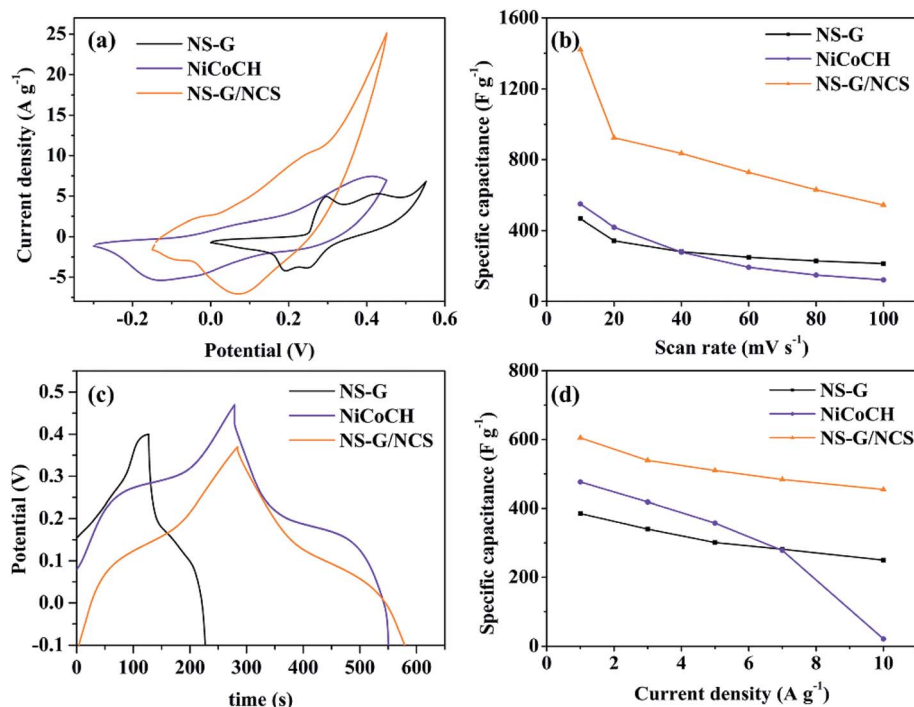


Fig. 5 CV curves of NS-G, NiCoCH and NS-G/NCS at 10 mV s⁻¹ (a); the specific capacitance as a function of scan rate from 10 to 100 mV s⁻¹ (b); GCD curves of NS-G, NiCoCH and NS-G/NCS at 1 A g⁻¹ (c); the specific capacitance as a function of current density from 1 to 10 A g⁻¹ (d).

with increasing scan rate due to the limitation in diffusion at high rate. Interestingly, NCS/NS-G electrode demonstrates the highest rate performance with specific capacitance of 500 F g⁻¹ at 100 mV s⁻¹, suggesting that the well uniform NCS nanoparticles embedded on NS-G surface can create the facile shortcut pathway maintaining such a charge/ion transport through faradaic reaction between active surface and electrolyte even at high current.

Furthermore, the comparative voltage–time curves of NS-G, NiCoCH and NS-G/NCS at 1 A g⁻¹ are presented in Fig. 5(c). GCD curves of all the as-prepared sample electrodes exhibit a very low potential drop at the initial of the discharge process and plateau curve of discharge profile, implying a low internal resistance in electrode system and a good pseudo-capacitance behavior.^{51,52} Among of them, NCS/NS-G electrode demonstrates the presence of the longest plateau discharge time, confirming the highest specific capacitance values. Apparently, the impregnated NCS nanoparticles on NS-G surface can enhance the electro-absorption and pseudocapacitance behavior, attributing to the synergistic effect of the well uniform NCS embedding on NS-G surface.²⁸ The calculated specific capacitance values from GCD results of the as-prepared electrode samples of NS-G, NiCoCH and NCS/NS-G are 385.22, 476.58 and 630.57 F g⁻¹ at 1 A g⁻¹, respectively, which is a good agreement with CV results. The further investigation on the rate performance of the as-prepared electrode samples at current densities ranging from 1 to 10 A g⁻¹ is shown in Fig. 5(d). Obviously, the co-presence of NCS and NS-G can improve the capacitive rate performance, resulting from the enhancing of ion diffusion contribution participating in charge storage

mechanism. Thus, this as-prepared electrode undoubtedly possesses the highest rate performance by maintaining 76.7% capacitance value from 1 to 10 A g⁻¹, which is superior to that of the other electrodes.

To further emphasize the influence of NS-G on charge storage mechanism of NS-G/NCS, the CV analysis of the as-prepared samples at various scan rates ranging from 10 to 50 mV s⁻¹ is examined. As seen in Fig. 6(a)–(c), CV curve of NS-G electrode reveals the rectangular-like shape with the presence of redox region, which preserves the reversible redox reaction even at high rate. In contrast, CV curve of NiCoCH electrode exhibits the reversible redox only at low scan rate. Obviously, the current response peak at oxidation and redox reaction is disappeared at over 30 mV s⁻¹ due to ion diffusion limitation of NiCoCH surface. In addition, the embedding NCS onto NS-G surface can enhance the reversible redox reaction of NCS as a presence of a pair redox reaction at high rate. The charge storage contribution of each electrode is analyzed using the relationship between the peak currents (i) and scan rate (ν) by plotting $i/\nu^{1/2}$ and $\nu^{1/2}$ (eqn (5)).

$$i = k_1\nu + k_2\nu^{1/2} \quad (5)$$

where k_1 and k_2 are slope at y-axis interception. The calculated $k_2\nu^{1/2}$ is represented to ionic diffusion contribution and $k_1\nu$ is referred to capacitive contribution.⁵³ Fig. 6(d)–(f) demonstrates the charge storage contribution fraction of NS-G, NiCoCH and NS-G/NCS electrodes with difference scan rates, respectively. Apparently, the ratio of ion diffusion/capacitive contribution in charge mechanism is related to the material characteristics. NS-



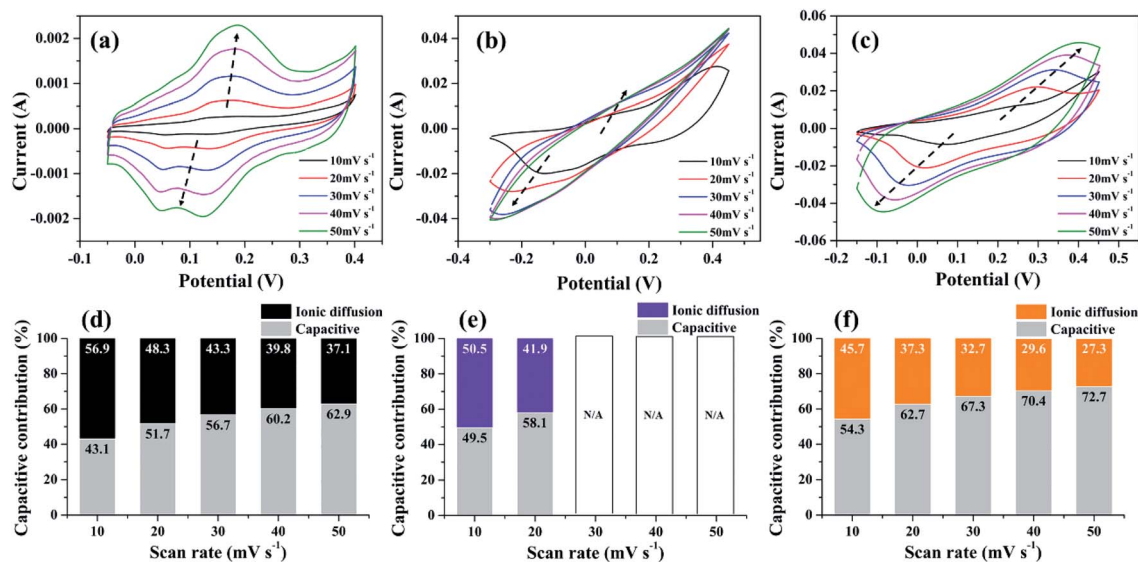


Fig. 6 CV curves at different scan rate from 10 to 50 mV s⁻¹ of NS-G (a); NiCoCH (b); NS-G/NCS (c); capacitive and ionic diffusion contributions at different scan rate of NS-G (d); NiCoCH (e); NS-G/NCS (f).

G, classified as EDLC, can well perform in ion diffusion control better than that of pseudocapacitive NiCoCH. It is clearly seen that the presence of NCS embedded onto NS-G surface can improve the reversible redox reaction of NCS, maintaining such ion/charge diffusion and capacitive behaviors on surface composites at high rate. The capacitive contributions of NS-G/NCS are 54.3%, 62.7%, 67.3%, 70.4% and 72.7% at 10, 20, 30, 40 and 50 mV s⁻¹, respectively. These fraction of capacitive

contribution of NS-G/NCS electrode suggests the improved rate capacitive behavior in electrode composites. As evidenced support by SEM micrograph, the well uniform distribution of NCS embedded on NS-G surface as framework structure on composite surfaces, which is greatly improved by our process, offer the utilization for active sites participating in charge/ion transport the excellent electrochemical reversibility and low internal resistance. These features of can be identified as: (i) the

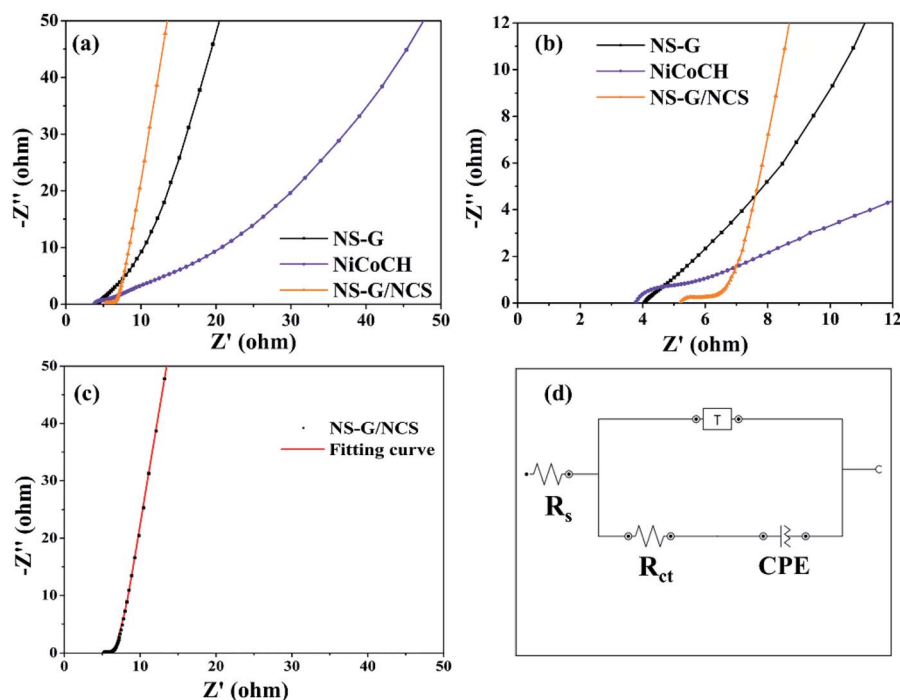


Fig. 7 Nyquist plots at frequency range 0.01–10 000 Hz of NS-G, NiCoCH and NS-G/NCS electrode (a and b); fitting curve of NS-G/NCS electrode (c); equivalent circuit diagram used for fitting EIS data (d).



Table 2 Equivalent circuit parameters of EIS of NS-G/NCS electrode

Sample	R_s (ohm)	R_{ct} (ohm)	Constant phase element (CPE; Q)		Tangent hyperbolic (T)		Chi square
			Y_0 (mMho s^N)	N	Y_0 (mMho $s^{(1/2)}$)	B ($s^{(1/2)}$)	
NS-G	3.86	1.11	3.8	0.729	9.07	0.45	0.07
NiCoCH	3.80	5.40	47.8	0.437	6.85	3.33	0.16
NS-G/NCS	5.20	1.59	61.0	0.904	11.50	1.12	0.02

synergistic effect of electric double layer (NS-G) and pseudo-capacitance behavior (NCS) in composite materials; (ii) the well uniform distribution of NCS nanoparticle embedded on NS-G surface as framework structure on composite surface; (iii) a large number of high electroactive surface sites from a nanoparticulate topography NCS on NS-G framework structure.

The comparative Nyquist plots of NS-G, NiCoCH and NS-G/NCS electrode samples using electrochemical impedance spectroscopic (EIS) at frequency range 0.01–10 000 Hz in 1 M KOH are investigated as shown in Fig. 7. The solution resistance (R_s), faradaic charge transfer resistance (R_{ct}) and the capacitive behavior of electrode materials are observed. All electrode samples expose two parts, consisting of a semicircular region at high/medium frequency and a relatively linear response at low frequency. The initial value of semicircle region at beginning plot refers to the resistance between material surface interaction between active materials, current corrector and electrolyte. The arc line intersection of semicircle corresponds with the total resistance of electrode, relating to electron-transfer-limited processes. A phase angle of relative linear region corresponds with ion/charge diffusion into porous and inter-layer of materials.^{54,55} Among of all the as-prepared electrode samples, NS-G/NCS electrode obviously demonstrates the low electronic/ionic resistances at high frequency and the smallest semicircle of arc line intersection curve with the vertical line characteristic of capacitive behavior at medium-low

frequencies, indicating the fastest ion/charge diffusion and lowest total resistance of the electrode behaviors. This result confirms the improvement of electrical conductivity and ion/charge transport of NS-G/NCS electrode, affecting from the well uniform NCS nanoparticles embedded on NS-G surface. In addition, the equivalent-circuit model is employed to simulate the EIS fitting spectra as seen in Fig. 7(d). The circuit fitting values and the calculated chi-square statistics are presented as in Table 2, confirming the lowest solution resistance (R_s) and charge transfer resistance (R_{ct}) of NS-G/NCS electrode (chi square of 0.02).

The cycle stability of samples is performed using symmetrical cell of CR2032 battery call at 0.1 A g^{-1} , employing from 1 to 10 000 cycles in 6 M KOH electrolyte. Fig. 8 demonstrates the comparative capacitance retention of all electrode samples from 1 to 10 000 cycles. It is clearly seen that NS-G can preserve the constant 100% stability after 10 000 cycles. Meanwhile, NiCoCH exhibits the activation of the phase-crystal-structure development during charge–discharge process, providing the retention increase to 131.2% after 10 000 cycles. For NCS/NS-G electrode, the capacitive performance is increased at initial 2400 cycles (136.2% retention) and is slightly decreased to 110% retention after 10 000 cycles. The increase in % retention during the charge/discharge cycling could be explained as the development of metastable structural state of transition metal oxide into crystal structure.^{56,57} This result implies the phase-crystal-structure development, resulting from surface liquidation of

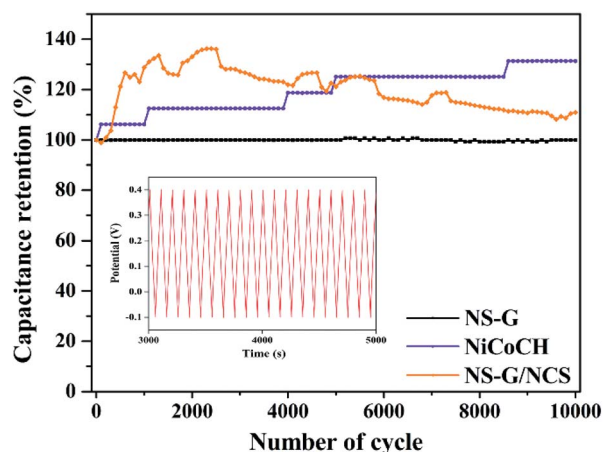


Fig. 8 Specific capacitance retention of NS-G, NiCoCH and NS-G/NCS as a function of cycle number at 0.1 A g^{-1} ; inset shows the GCD curve of NS-G/NCS electrode.

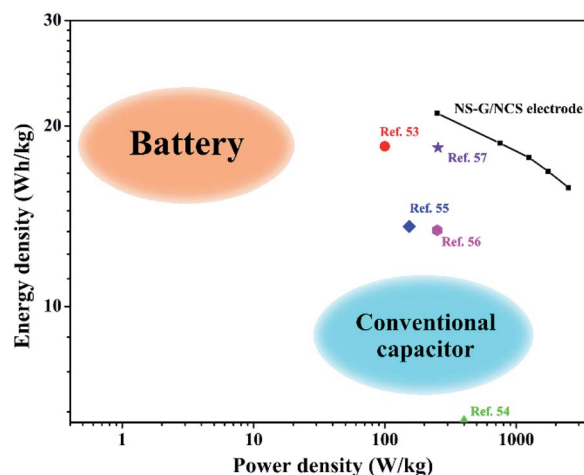


Fig. 9 Ragone chart of NS-G/NCS.



Table 3 The comparative electrochemical performance of NS-G/NCS with other reports

Materials	Electrolyte	Specific capacitance	Cycling stability	References
Ni ₃ S ₂ @VO ₂	2 M KOH	1.09C cm ⁻² (1 mA cm ⁻²)	91.7%, 1000 cycles	64
Ni ₃ S ₂ films	1 M KOH	3.42 F cm ⁻² (1 mA cm ⁻²)	76.6%, 2000 cycles	54
S-doped ZnCo ₂ O ₄ micro-spindles	3 M KOH	522 F g ⁻¹ (0.5 A g ⁻¹)	78.0%, 5000 cycles	65
PANI/NiHCF/sCNT	1 M H ₂ SO ₄	430.8 F g ⁻¹ (2 mA cm ⁻²)	74.3%, 1000 cycles	66
Ni-doped CoHCF microcubes	1 M Na ₂ SO ₄	466 F g ⁻¹ (5 mV s ⁻¹)	87.9%, 2000 cycles	67
Co ₃ O ₄ nanotubes	6 M KOH	574 F g ⁻¹ (0.1 A g ⁻¹)	95.0%, 1000 cycles	31
Co ₃ O ₄ thin film	1 M KOH	162 F g ⁻¹ (10 mV s ⁻¹)	98.3%, 1000 cycles	60
Nanorod nickel cobalt hydroxide	1 M KOH	456 F g ⁻¹ (20 mV s ⁻¹)	91.0%, 1000 cycles	68
Graphene/Co ₃ O ₄ /polypyrrole	6 M KOH	385 F g ⁻¹ (1 A g ⁻¹)	—	62
NS-G/NCS	6 M KOH	1420.2 F g ⁻¹ (10 mV s ⁻¹)	110%, 10 000 cycles	This work

the large amount of sulfate species during charge–discharge process.^{43,58}

In addition, the energy-power performance of NCS/NS-G electrode is revealed as the Ragone plot in Fig. 9. The energy density and power density are calculated as following the eqn (2) and (3), respectively. NCS/NS-G electrode can reach an energy density in rang of 19.35–14.83 W h kg⁻¹ and power density in range of 235.0–2350.0 W kg⁻¹. The comparison of electrochemical NCS/NS-G electrode sample with other reports are also illustrated in Table 3. These results suggest a high sufficient performance of NCS/NS-G electrode and its performance outperform many previous reports of nickel/cobalt-based materials such as cobalt sulfide/graphitic carbon nitride (18.5 W h kg⁻¹ at power density of 99.8 W kg⁻¹),⁵⁹ Co₃O₄ thin film (6.4 W h kg⁻¹ at power density of 0.4 kW kg⁻¹),⁶⁰ beta phase Co(OH)₂ nanowires (13.6 W h kg⁻¹ at power density of 153 W kg⁻¹),⁶¹ graphene/Co₃O₄/polypyrrole (13.4 W h kg⁻¹ at power density of 250 W kg⁻¹),⁶² and nickel cobalt sulfide nanosheets on Ni nanowire film (18.4 mW h cm⁻³ at power density of 254.5 mW cm⁻³).⁶³

4. Conclusion

Nickel cobalt sulfide nanoparticle/nitrogen and sulfur dual doped graphene are successfully synthesized by the facile hydrothermal process. The significant development of crystal-phase-structure of sea-urchin-like NiCoCH through hydrothermal process in presence of graphene oxide and thiourea yields the well uniform nickel cobalt sulfide nanoparticles embedded on nitrogen and sulfur dual doped graphene surface. This unique structure feature offers the significantly enhanced active sites to promotes fast ion/charge transport and enhances the synergistic electrochemical performance. NCS/NS-G electrode composites can reach the high specific capacitance of 1420.2 F g⁻¹ at 10 mV s⁻¹ and 630.6 at 1 A g⁻¹. In the presence of NS-G, the capacitance rate (ranging from 1 to 10 A g⁻¹) of NCS/NS-G is improved by maintaining 76.65% initial capacitive values. NCS/NS-G symmetrical supercapacitors CR2302 cell exhibits the excellent cycling stability (110% after 10 000 cycles). The high energy density of 19.35–14.83 W h kg⁻¹ and high-power density of 235.0–2350.0 W kg⁻¹ evidently confirm the potential usage of the developed materials as a candidate for high performance supercapacitors. The authors believe that the

novel route with low-cost effectiveness in this study can open new avenue for scale-up process of energy storage materials.

Author contributions

Nutthapong Poompiew: methodology, formal analysis, investigation, visualization, writing – original draft. Prasit Pattanauwat: conceptualization, writing – review and editing, supervision. Pranut Potiyaraj: conceptualization, writing – review and editing, project administration.

Conflicts of interest

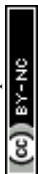
There are no conflicts to declare.

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References

- B. Dunn, H. Kamath and J.-M. Tarascon, *Science*, 2011, **334**, 928.
- D. P. Dubal, O. Ayyad, V. Ruiz and P. Gomez-Romero, *Chem. Soc. Rev.*, 2015, **44**, 1777–1790.
- S. Najib and E. Erdem, *Nanoscale Adv.*, 2019, **1**, 2817–2827.
- L. Wu, L. Shen, T. Wang, X. Xu, Y. Sun, Y. Wang, Y. Zhao, Y. Du and W. Zhong, *J. Alloys Compd.*, 2018, **766**, 527–535.
- M. I. Rafiq, T. Farid, J. Zhou, A. Ali, J. Tang and W. Tang, *J. Alloys Compd.*, 2019, **811**, 151858.
- L. Li, H. Song, Q. Zhang, J. Yao and X. Chen, *J. Power Sources*, 2009, **187**, 268–274.
- J. Zhu, M. Chen, H. Qu, X. Zhang, H. Wei, Z. Luo, H. A. Colorado, S. Wei and Z. Guo, *Polymer*, 2012, **53**, 5953–5964.
- P. Pattanauwat and D. Aht-ong, *Electrochim. Acta*, 2017, **224**, 149–160.



- 9 P. Pattanauwat and D. Aht-Ong, *Mater. Lett.*, 2016, **184**, 60–64.
- 10 Y. He, P. Zhang, H. Huang, X. Li, X. Zhai, B. Chen and Z. Guo, *Electrochim. Acta*, 2020, **343**, 136140.
- 11 A. U. Ammar, I. D. Yildirim, F. Bakan and E. Erdem, *Beilstein J. Nanotechnol.*, 2021, **12**, 49–57.
- 12 S. Najib, F. Bakan, N. Abdullayeva, R. Bahariqushchi, S. Kasap, G. Franzo, M. Sankir, N. Demirci Sankir, S. Mirabella and E. Erdem, *Nanoscale*, 2020, **12**, 16162–16172.
- 13 M. Toufani, S. Kasap, A. Tufani, F. Bakan, S. Weber and E. Erdem, *Nanoscale*, 2020, **12**, 12790–12800.
- 14 S. Kasap, I. I. Kaya, S. Repp and E. Erdem, *Nanoscale Adv.*, 2019, **1**, 2586–2597.
- 15 D. Pan, M. Zhang, Y. Wang, Z. Yan, J. Jing and J. Xie, *Chem. Phys. Lett.*, 2017, **685**, 457–464.
- 16 P. Asen, S. Shahrokhian and A. Irajizad, *Int. J. Hydrogen Energy*, 2017, **42**, 21073–21085.
- 17 H. Su, T. Wang, S. Zhang, J. Song, C. Mao, H. Niu, B. Jin, J. Wu and Y. Tian, *Solid State Sci.*, 2012, **14**, 677–681.
- 18 Z. Zhou, N. Cai and Y. Zhou, *Mater. Chem. Phys.*, 2005, **94**, 371–375.
- 19 T. V. Nguyen, L. T. Son, P. M. Thao, L. T. Son, D. T. Phat, N. T. Lan, N. V. Nghia and T. V. Thu, *J. Alloys Compd.*, 2020, **831**, 154921.
- 20 Z. Ma, Z. Sun, H. Jiang, F. Li, Q. Wang and F. Qu, *Appl. Surf. Sci.*, 2020, **533**, 147521.
- 21 L. G. Beka, X. Li and W. Liu, *Sci. Rep.*, 2017, **7**, 2105.
- 22 T.-F. Hung, Z.-W. Yin, S. B. Betzler, W. Zheng, J. Yang and H. Zheng, *Chem. Eng. J.*, 2019, **367**, 115–122.
- 23 B. Chameh, M. Moradi and S. Kaveian, *Synth. Met.*, 2020, **260**, 116262.
- 24 Y. Chen, C. Jing, X. Fu, M. Shen, K. Li, X. Liu, H.-C. Yao, Y. Zhang and K. X. Yao, *Chem. Eng. J.*, 2020, **384**, 123367.
- 25 P. Miao, J. He, Z. Sang, F. Zhang, J. Guo, D. Su, X. Yan, X. Li and H. Ji, *J. Alloys Compd.*, 2018, **732**, 613–623.
- 26 Y. Liu, G. Jiang, B. Yu, U. Evariste and P. Ma, *J. Alloys Compd.*, 2020, **832**, 154909.
- 27 P.-R. Wu, C.-L. Wu and D.-H. Chen, *J. Alloys Compd.*, 2018, **735**, 409–416.
- 28 F. Y. Ban, S. Jayabal, H. N. Lim, H. W. Lee and N. M. Huang, *Ceram. Int.*, 2017, **43**, 20–27.
- 29 N. Parveen, M. O. Ansari, S. A. Ansari and M. H. Cho, *J. Mater. Chem. A*, 2016, **4**, 233–240.
- 30 J. Huo, P. Zheng, X. Wang and S. Guo, *Appl. Surf. Sci.*, 2018, **442**, 575–580.
- 31 J. Xu, L. Gao, J. Cao, W. Wang and Z. Chen, *Electrochim. Acta*, 2010, **56**, 732–736.
- 32 J. Zhang, F. Liu, J. P. Cheng and X. B. Zhang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 17630–17640.
- 33 P. Sivakumar, M. Jana, M. Kota, H. S. Lee and H. S. Park, *J. Alloys Compd.*, 2019, **781**, 515–523.
- 34 S. Li, K. Yang, P. Ye, H. Jiang, Z. Zhang, Q. Huang and L. Wang, *Appl. Surf. Sci.*, 2019, **473**, 326–333.
- 35 S. Wen, K. Qin, P. Liu, N. Zhao, C. Shi, L. Ma and E. Liu, *J. Alloys Compd.*, 2019, **783**, 625–632.
- 36 Z. Li, M. Tian, X. Sun, H. Zhao, S. Zhu and X. Zhang, *J. Alloys Compd.*, 2019, **782**, 986–994.
- 37 A. G. Kannan, J. Zhao, S. G. Jo, Y. S. Kang and D.-W. Kim, *J. Mater. Chem. A*, 2014, **2**, 12232–12239.
- 38 C. Zhao, C. Chen, F. Du and J. Wang, *RSC Adv.*, 2015, **5**, 38533–38537.
- 39 D. Khalafallah, Q. Zou, M. Zhi and Z. Hong, *Electrochim. Acta*, 2020, **350**, 136399.
- 40 H. Wang, C. M. B. Holt, Z. Li, X. Tan, B. S. Amirkhiz, Z. Xu, B. C. Olsen, T. Stephenson and D. Mitlin, *Nano Res.*, 2012, **5**, 605–617.
- 41 J. Cao, Y. Hu, Y. Zhu, H. Cao, M. Fan, C. Huang, K. Shu, M. He and H. C. Chen, *Chem. Eng. J.*, 2021, **405**, 126928.
- 42 M. Gao, L. He, Z. Y. Guo, Y. R. Yuan and W. W. Li, *ACS Appl. Mater. Interfaces*, 2020, **12**, 443–450.
- 43 M. Gao, Z. Y. Guo, X. Y. Wang and W. W. Li, *ChemSusChem*, 2019, **12**, 5291–5299.
- 44 Z. Wei, J. Yuan, S. Tang, D. Wu and L. Wu, *J. Colloid Interface Sci.*, 2019, **542**, 15–22.
- 45 C.-H. Mun, C. V. V. Muralee Gopi, R. Vinodh, S. Sambasivam, I. M. Obaidat and H.-J. Kim, *J. Energy Storage Convers.*, 2019, **26**, 100925.
- 46 A. Bera, A. Maitra, A. K. Das, L. Halder, S. Paria, S. K. Si, A. De, S. Ojha and B. B. Khatua, *ACS Appl. Electron. Mater.*, 2020, **2**, 177–185.
- 47 X. Cao, Y. Liu, Y. Zhong, L. Cui, A. Zhang, J. M. Razal, W. Yang and J. Liu, *J. Mater. Chem. A*, 2020, **8**, 1837–1848.
- 48 M. Dong, Z. Wang, G. Yan, J. Wang, H. Guo and X. Li, *J. Alloys Compd.*, 2020, **822**, 153645.
- 49 Y. Zou, C. Cai, C. Xiang, P. Huang, H. Chu, Z. She, F. Xu, L. Sun and H.-B. Kraatz, *Electrochim. Acta*, 2018, **261**, 537–547.
- 50 X. Ren, Y. Du, M. Song, Y. Chen, Y. Zhou, F. Ma and J. Wan, *J. Alloys Compd.*, 2019, **806**, 1481–1490.
- 51 Z. Wang, H. Wang, S. Ji, H. Wang, D. J. L. Brett and R. Wang, *J. Alloys Compd.*, 2020, **814**, 151789.
- 52 W. Hu, H. Wei, Y. She, X. Tang, M. Zhou, Z. Zang, J. Du, C. Gao, Y. Guo and D. Bao, *J. Alloys Compd.*, 2017, **708**, 146–153.
- 53 Y. Lan, H. Zhao, Y. Zong, X. Li, Y. Sun, J. Feng, Y. Wang, X. Zheng and Y. Du, *Nanoscale*, 2018, **10**, 11775–11781.
- 54 J. Li, S. Wang, T. Xiao, X. Tan, P. Xiang, L. Jiang, C. Deng, W. Li and M. Li, *Appl. Surf. Sci.*, 2017, **420**, 919–926.
- 55 D. Y. Kim, G. S. Ghodake, N. C. Maile, A. A. Kadam, D. Sung Lee, V. J. Fulari and S. K. Shinde, *Sci. Rep.*, 2017, **7**, 9764.
- 56 P. R. Martins, A. L. Araújo Parussulo, S. H. Toma, M. A. Rocha, H. E. Toma and K. Araki, *J. Power Sources*, 2012, **218**, 1–4.
- 57 T. Deng, W. Zhang, O. Arcelus, J. G. Kim, J. Carrasco, S. J. Yoo, W. Zheng, J. Wang, H. Tian, H. Zhang, X. Cui and T. Rojo, *Nat. Commun.*, 2017, **8**, 15194.
- 58 F. I. Dar, K. R. Moonosawmy and M. Es-Souni, *Nanoscale Res. Lett.*, 2013, **8**, 363.
- 59 D. Jiang, Q. Xu, S. Meng, C. Xia and M. Chen, *J. Alloys Compd.*, 2017, **706**, 41–47.
- 60 S. G. Kandalkar, J. L. Gunjekar and C. D. Lokhande, *Appl. Surf. Sci.*, 2008, **254**, 5540–5544.



- 61 Y. Tang, Y. Liu, S. Yu, S. Mu, S. Xiao, Y. Zhao and F. Gao, *J. Power Sources*, 2014, **256**, 160–169.
- 62 M. Khalaj, A. Sedghi, H. N. Miankushki and S. Z. Golkhatmi, *Energy*, 2019, **188**, 116088.
- 63 S. Xu, C. Su, T. Wang, Y. Ma, J. Hu, J. Hu, N. Hu, Y. Su, Y. Zhang and Z. Yang, *Electrochim. Acta*, 2018, **259**, 617–625.
- 64 H. Du, F. Ding, J. Zhao, X. Zhang, Y. Li, Y. Zhang, J. Li, X. Yang, K. Li and Y. Yang, *Appl. Surf. Sci.*, 2020, **508**, 144876.
- 65 Y. Yang, C. Yang, K. Tao, Q. Ma and L. Han, *Vacuum*, 2020, **181**, 109740.
- 66 Y. Wang, Y. Yang, X. Zhang, C. Liu and X. Hao, *J. Solid State Electrochem.*, 2015, **19**, 3157–3168.
- 67 Y. Qiu, Y. Lin, H. Yang and L. Wang, *J. Alloys Compd.*, 2019, **806**, 1315–1322.
- 68 R. R. Salunkhe, K. Jang, S.-w. Lee and H. Ahn, *RSC Adv.*, 2012, **2**, 3190.

