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Hydrogen solubility and diffusivity at $\Sigma 3$ grain boundary of PdCu[†]

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First principles calculations have been performed to comparatively reveal hydrogen solubility and diffusivity at grain boundaries of BCC and FCC PdCu phases. It is found that the temperature-dependent hydrogen solubility at BCC $\Sigma 3$ (112) GB of PdCu seems much higher than that in BCC PdCu bulk, while hydrogen solubility in FCC $\Sigma 3$ (111) GB of PdCu is much lower than that in its corresponding FCC bulk. Calculations also reveal that grain boundary has an important effect on hydrogen diffusion of BCC and FCC PdCu, *i.e.*, hydrogen diffusivities of BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) grain boundaries of PdCu seem much smaller and bigger than those of its corresponding bulks, respectively. The predicted results could deepen the comprehension of hydrogen solubility and diffusion of PdCu phases.

1. Introduction

Hydrogen solution and diffusion at grain boundaries of various metals have raised great research interest in the past decades due to the hydrogen embrittlement, intergranular cracking, increased creep rates, and decreased fatigue life *etc.*, while the main conclusions available in the literature are not consistent with each other.^{1–13} For instance, hydrogen diffusion in grain boundaries of nickel was reported to be much faster than that in the bulk, which would be fundamentally due to a lower activation energy of diffusion in the area of the grain.^{3,6} On the other hand, the effects of grain boundary on hydrogen solution and diffusion are related to the grain size and the type of grain boundary, *e.g.*, hydrogen diffusivity in the grain boundary of aluminum and nickel with fine grains is smaller than that in the bulks.^{7,10,11,13} Nevertheless, some researchers also reported that grain boundary has no effect on the solution or diffusion of hydrogen.^{7,9}

As to the hydrogen permeated Pd membranes, Mütschele and Kirchheim found that hydrogen solubility in nanocrystalline Pd is much bigger than that in its single crystalline counterpart.¹⁴ With the decrease of the grain size, hydrogen solution in the α phase of PdH becomes bigger, whereas smaller in the β phase of PdH.^{14,15} Interestingly, Stuhr *et al.* made a quite different conclusion that compared with coarse-grained Pd, no change of hydrogen solubility was discovered for nanocrystalline Pd.¹⁶ In addition, hydrogen diffusion in grain boundaries of Pd seems much faster than that within the grains,⁸ and is also

found to be concentration dependent, *i.e.*, hydrogen diffusion coefficients are lower than single crystalline value at low hydrogen contents, and *vice versa* for higher contents.¹⁴

It is well known that the PdCu membrane is a kind of excellent candidates for hydrogen separation and purification, and compared with pure Pd and other Pd alloys, PdCu possesses excellent hydrogen selectivity, high thermal stability, superior resistance against poisoning, moderate mechanical properties, relatively low price, and wide operating temperature range.^{17–20} Regarding hydrogen solubility and hydrogen diffusivity of various PdCu phases, there are already a lot of experimental and theoretical investigations in the literature.^{17,19–43} These experimental studies are mainly concentrated on polycrystalline phases of PdCu,^{19–22,24–35,37,39} while the reported theoretical calculations are all related to single crystals of PdCu.^{17,23,36,38,40–42} In other words, the effects of grain boundary on hydrogen behaviors of PdCu membranes need further research investigations.

By means of highly accurate first principles calculations based on density functional theory,^{43–47} the present study is aimed to have a comparative investigation of hydrogen solubility and diffusivity at grain boundaries of BCC and FCC PdCu. Both BCC and FCC structures of PdCu are intentionally selected as they are two main phases in the Pd–Cu phase diagram.⁴³ Specifically, the PdCu phases with the stoichiometric ratio of 1 : 1 are intentionally selected due to their superior performance of H permeability.^{18,35,37,38} In addition, a lot of investigations have found that the $\Sigma 3$ (112) grain boundary of BCC metals has the lowest grain boundary energy,^{48–52} and the $\Sigma 3$ (111) grain boundaries of FCC metals are energetically stable structure with the lowest grain boundary energy in the literature.^{13,51,53} Accordingly, the $\Sigma 3$ (112) grain boundaries of BCC PdCu and $\Sigma 3$ (111) of FCC PdCu are therefore chosen in the present study. And the hydrogen

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solubility and hydrogen diffusivity of PdCu bulks³⁸ are used for the sake of comparison. The derived results will be compared with available experimental and theoretical evidence in the literature, and could provide a deep understanding of the effects of grain boundary on hydrogen solubility and diffusivity of BCC and FCC PdCu.

2. Method of calculation

The present first principles calculations are implemented based on density functional theory with the projector-augmented wave (PAW) method using Vienna *Ab initio* Simulation Package (VASP).^{44,45,48,49} The generalized gradient approximation (GGA) of Perdew–Burke–Erzerof (PBE) is selected to describe exchange-correlation function,⁵⁰ and the cutoff energy of plane wave basis is 400 eV. The temperature-smearing method of Methfessel–Paxton⁵¹ and the tetrahedron method with the Blöchl corrections⁵² are adopted for structure relaxation and static calculations, respectively. The convergence criteria of relaxation and static calculations are 10^{-5} and 10^{-6} eV, respectively. The conjugate gradient method is used to minimize the Hellmann–Feynman forces in the ionic relaxations with the force stopping criterion of $0.02 \text{ eV } \text{Å}^{-1}$.

Accordingly, the atomic composition of Pd₅₀Cu₅₀ is purposely chosen as a result of its superior hydrogen permeability.^{18,35,37,38} The grain boundaries of BCC $\Sigma 3$ (112) (48 atoms) and FCC $\Sigma 3$ (111) (48 atoms) of PdCu are constructed based on coincidence site lattice theory (CSL),⁵³ and are shown vividly in Fig. 1. It should be noted that the CSL and structural-unit (SU) models yield the same structure and energy for the $\Sigma 3$ (112) or $\Sigma 3$ (111) grain boundary.⁵³ The lattice constants of BCC PdCu (3.026 Å) and FCC PdCu (3.817 Å) obtained recently³⁸ are used in the present study. After the test calculations, the dimensions of BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) of PdCu are $5.241 \times 4.279 \times 29.769 \text{ Å}^3$ and $5.398 \times 4.675 \times 26.445 \text{ Å}^3$, respectively. To simulate hydrogen solubility and hydrogen diffusivity, one hydrogen atom is added at the octahedral (O) and tetrahedral (T) sites of each supercell. It should be pointed out that the O1 site is surrounded by four Pd atoms and two Cu atoms, and the O2 site by two Pd and four Cu atoms. During each calculation, periodic boundary conditions are added to the model in three directions. To determine the GB structure with the lowest energy, multiple initial configurations are sampled with a series of rigid body translations (RBTs) in the direction perpendicular to the GB. After the test calculations, the *k*-meshes of $7 \times 7 \times 1$ and $9 \times 9 \times 1$ are chosen for relaxation and static calculations, respectively.

For the calculation of H diffusion in BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) of PdCu, the climbing image nudged elastic band (CI-NEB) method⁵⁴ is used to search the minimum energy paths. The spring constant and force convergence criterion are 5.0 and $0.05 \text{ eV } \text{Å}^{-1}$, respectively. The frequency calculations are performed for the diffusion paths to verify the saddle points with only one imaginary frequency, and the energy barrier is derived as the difference between the total energy of saddle point and initial configuration.

3. Results and discussion

3.1. Hydrogen solubility at $\Sigma 3$ grain boundary of PdCu

A lot of researches have shown that the hydrogen solubility of FCC PdCu phase is much higher than that of BCC counterpart, which is due to the stronger binding of hydrogen at FCC PdCu phase.^{20,34,38,55} As to the effect of grain boundary (GB) on hydrogen solubility at PdCu phases, however, there is not any report so far in the literature. The present first principles calculation is therefore executed to have a comparative study of fundamental influence of GB on hydrogen solubility of PdCu phases.

Accordingly, the binding energy (E_b) and zero point energy (ZPE) of hydrogen at $\Sigma 3$ GB of PdCu could be calculated as follows:^{38,40,42}

$$E_b = E_{\text{PdCuH}}^{\text{GB}} - E_{\text{PdCu}}^{\text{GB}} - \frac{1}{2}E_{\text{H}_2}, \quad (1)$$

$$E_{\text{ZPE}} = \frac{1}{2} \sum_i h\nu_i, \quad (2)$$

where E_{PdCuH} and E_{PdCu} are total energies of GB of PdCu with and without one hydrogen, respectively; E_{H_2} is total energy of a H₂ molecule; ν_i is the vibration frequency of interstitial H atom in GB, and is obtained using the harmonic approximation through fixing metal atoms due to their much bigger mass than H atom.

The segregation energy (E_{seg}) of hydrogen at GB of PdCu is then calculated according to the following formula:

$$E_{\text{seg}} = E_b^{\text{GB}} - E_b^{\text{bulk}}, \quad (3)$$

where E_b^{GB} and E_b^{bulk} are binding energies of hydrogen at GB and bulk of PdCu, respectively. Fundamentally, the segregation energy is regarded as the efficiency of GB to trap H atoms, *i.e.*, the negative E_{seg} indicates H atoms are easier to gather in GB, and *vice versa*.

The hydrogen solubility (C) in GB of PdCu could be then derived by the Sieverts' law:

$$C = K_s \sqrt{P}, \quad (4)$$

where P is pressure set as the standard atmospheric pressure (1 atm), and K_s is the Sieverts' constant:⁵⁵

$$K_s = \frac{\exp\left(\beta \left[\sum_i \frac{h\gamma_i}{4} - E_b - \frac{1}{2} \sum_i h\nu_i \right]\right)}{\sqrt{\alpha} \sqrt{1 - \exp\left(-\beta \sum_i \frac{h\gamma_i}{2}\right)} \prod_i (1 - e^{-\beta h\nu_i})}, \quad (5)$$

$$\alpha = \left(\frac{2\pi m k_B T}{\hbar^2}\right)^{3/2} \frac{4\pi^2 I (k_B T)^2}{\hbar^2}, \quad (6)$$

$$\beta = \frac{1}{k_B T}, \quad (7)$$



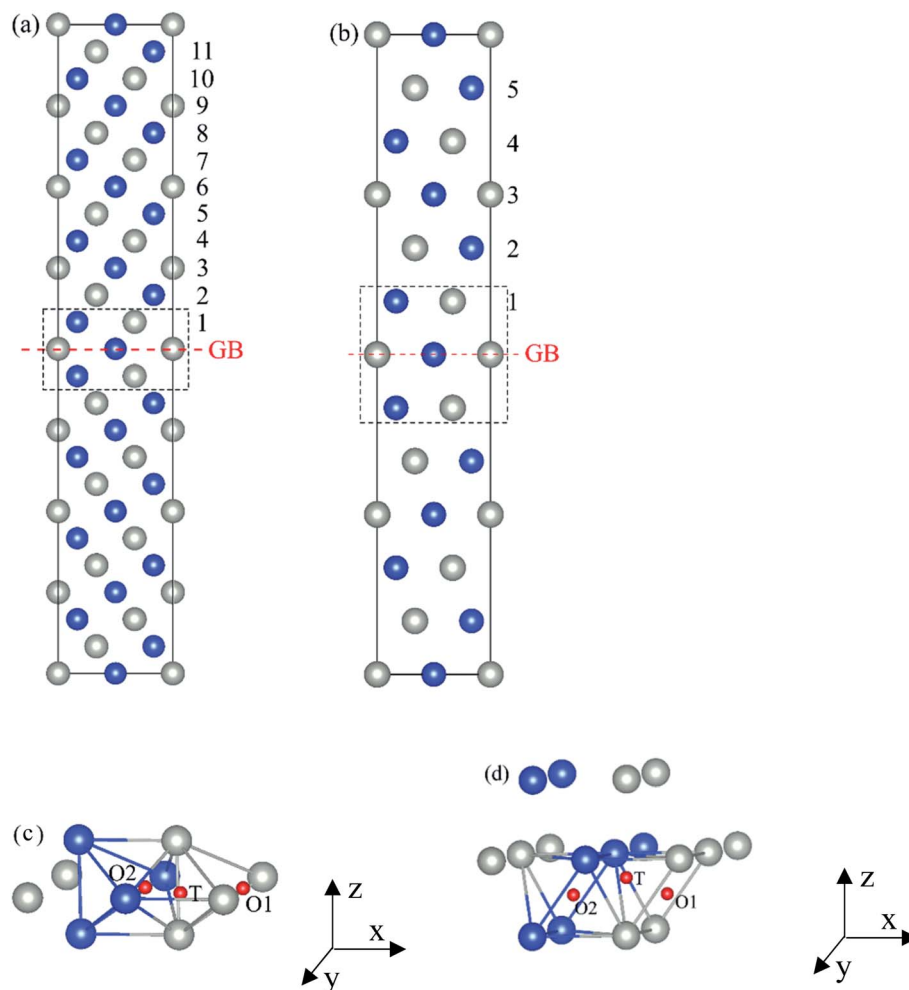


Fig. 1 Atomic structure of (a) BCC $\Sigma 3$ (112) and (b) FCC $\Sigma 3$ (111) grain boundary (GB) of PdCu (c) and (d) are octahedral (O) and tetrahedral (T) sites of H atom in BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GB, respectively. The x-axis is $[111]_{\text{BCC PdCu}}$ and $[011]_{\text{FCC PdCu}}$, the y-axis is $[110]_{\text{BCC PdCu}}$ and $[110]_{\text{FCC PdCu}}$, the z-axis is $[112]_{\text{BCC PdCu}}$ and $[111]_{\text{FCC PdCu}}$. The gray, blue and red spheres represent Pd, Cu and H atoms, respectively.

where γ_i is the vibration frequency of gaseous H_2 , m and I are the mass and molecular moment of inertia of H_2 , respectively; ν_i is the vibration frequency of interstitial H atom in GB; k_B is the Boltzmann constant and \hbar is the Planck constant. It should be pointed out that spin polarization is used to calculate the

energy, vibration frequency, and molecular moment of inertia of H_2 molecule.

Consequently, the E_b , ZPE, and E_{seg} values of hydrogen in BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GB of PdCu are obtained and summarized in Table 1. In addition, the temperature-dependent hydrogen solubility of BCC $\Sigma 3$ (112) and FCC $\Sigma 3$

Table 1 Binding energy (E_b) and segregation energy (E_{seg}) of H atom at octahedral (O1 and O2) and tetrahedral (T) interstitial sites of BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) grain boundaries of PdCu. The relevant zero point energy (ZPE) is added in the parenthesis after E_b . The corresponding E_b values of H in PdCu bulk³⁸ are also listed for comparison

Site	H in BCC PdCu			H in FCC PdCu		
	$\Sigma 3$ (112)	Bulk ³⁸	E_{seg} (eV)	$\Sigma 3$ (111)	Bulk ³⁸	E_{seg} (eV)
O1	-0.172 (0.133)	-0.029 (0.138)	-0.136	-0.042 (0.154)	-0.069 (0.089)	0.036
O2	-0.039 (0.127)	0.060 (0.120)	-0.085	0.149 (0.149)	0.136 (0.070)	0.013
T	-0.157 (0.211)	-0.059 (0.192)	-0.119	0.193 (0.205)	0.160 (0.155)	0.04



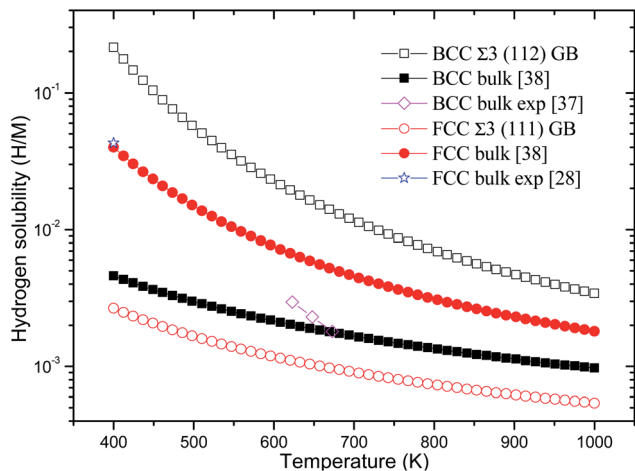


Fig. 2 Temperature-dependent hydrogen solubility at BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GB of PdCu. The corresponding values in BCC and FCC PdCu bulks³⁸ are also listed for comparison. H and M represent hydrogen and metal atoms, respectively.

(111) GB of PdCu is also derived and shown in Fig. 2. The corresponding binding energies and ZPE of H in PdCu bulks³⁸ are also listed in Table 1, and hydrogen solubility of PdCu bulks³⁸ is included in Fig. 2 for the sake of comparison. Several characteristics could be discerned from Table 1 and Fig. 2.

Firstly, the binding energy (E_b) of H in each interstitial site (O1, O2, and T) of BCC $\Sigma 3$ (112) GB of PdCu is lower than the corresponding value in BCC PdCu bulk,³⁸ suggesting that H should be energetically more favorable in BCC $\Sigma 3$ (112) GB of PdCu. Consequently, the hydrogen solubility at BCC $\Sigma 3$ (112) GB of PdCu seems much higher than that in BCC PdCu bulk.³⁸ That is to say, hydrogen energetically prefers to stay at the GB area of BCC PdCu, rather than within the grain. The calculated hydrogen solubility of BCC PdCu bulk³⁸ is slightly lower than the corresponding experimental measurement,³⁷ probably due to the higher hydrogen solubility at the BCC PdCu $\Sigma 3$ (112) GB. Such a theoretical prediction about hydrogen solubility at BCC

$\Sigma 3$ (112) GB of PdCu from the present study is similar to the experimental observations regarding hydrogen solubility in nanocrystalline Pd and the α phase of Pd.^{14,15}

Secondly, for each interstitial site of O1, O2, and T, FCC $\Sigma 3$ (111) GB of PdCu possesses higher binding energy of hydrogen than its bulk. This comparison signifies that hydrogen would be thermodynamically unfavorable in FCC $\Sigma 3$ (111) GB of PdCu. As shown in Fig. 2, the curve of temperature-dependent hydrogen solubility in FCC $\Sigma 3$ (111) GB of PdCu is therefore much lower than that in its corresponding FCC bulk. In other words, GB has an important effect to reduce hydrogen solubility of FCC PdCu, which is consistent with similar experimental discovery of the β phase of Pd¹⁵ as well as the theoretical prediction of Ni GB.¹³ It should be noted that the ZPE values of H at BCC $\Sigma 3$ (112) GB are close to that in bulk, while the ZEP values at FCC $\Sigma 3$ (111) GB are higher than that in the bulk.

Thirdly, the E_{seg} values of H at the O1, O2, and T sites of BCC $\Sigma 3$ (112) GB of PdCu are big and negative values of -0.136 , -0.085 , and -0.119 eV, respectively, indicating that hydrogen has a strong tendency to segregate in the GB area of BCC PdCu. Such a theoretical prediction from the present study is compatible with similar experimental observations of Pd.¹⁶ On the contrary, hydrogen seems difficult to segregate to the GB area of FCC PdCu due to its slightly positive E_{seg} values. Interestingly, the above drastic comparison regarding hydrogen segregation in GB of BCC and FCC PdCu is very similar to that reported about $\Sigma 3$ grain boundary of Fe in the literature.⁵⁶ It should be pointed out that thermal expansion of the lattice and vacancy would have some effects on H binding energy and hydrogen solubility of PdCu, and further studies are welcome to find out these effects.

To have a deep understanding of the effects of grain boundary on hydrogen solubility of PdCu, the electronic structures of H at various interstitial sites of PdCu are calculated and compared with each other. As a typical example, Fig. 3 displays the total density of states of H atom at the T site of BCC and FCC $\Sigma 3$ GBs and bulk of PdCu. It can be seen from Fig. 3 that the DOSs of H atom BCC $\Sigma 3$ (112) GB below the Fermi level (E_f) are

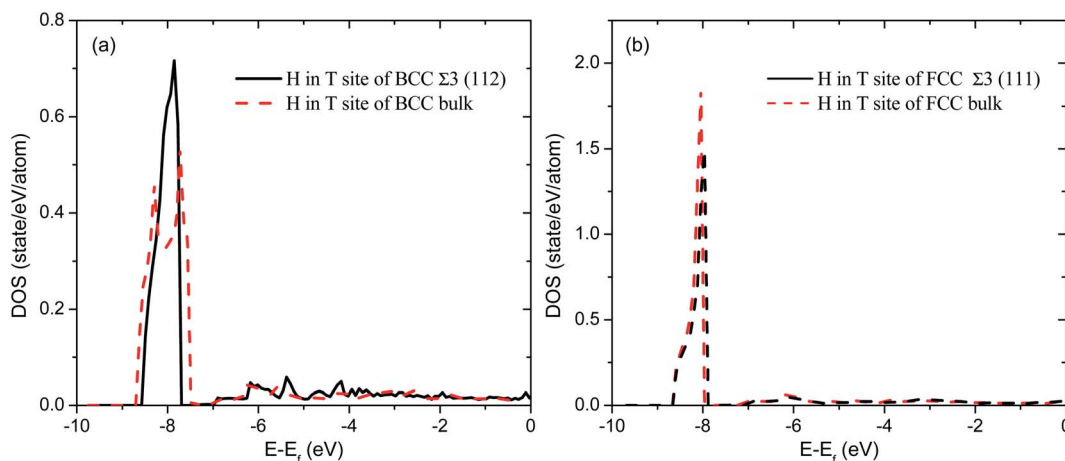


Fig. 3 Comparison of total densities of states (DOS) of H atom at the T site of (a) $\Sigma 3$ (112) GB and bulk of BCC PdCu (b) $\Sigma 3$ (111) GB and bulk of FCC PdCu.



more centralized than those in BCC PdCu bulk. In addition, the DOS peak value of H atom at BCC $\Sigma 3$ (112) GB of PdCu is 0.716 states per eV per atom, which is higher than the corresponding value of 0.535 states per eV per atom in BCC PdCu bulk. Similarly, the H atom at the T site of FCC $\Sigma 3$ (111) GB has a DOS peak of 1.49 states per eV per atom below E_f , and this DOS peak is smaller than the corresponding value (1.85 states per eV per atom) of H atom in FCC PdCu bulk. By means of the Bader analysis, the charge of H atom at the T site of BCC $\Sigma 3$ (112) GB of PdCu is derived to be 1.15, which is bigger than the value of 1.108 for the corresponding PdCu bulk. The above characteristics of electronic structures indicate that H atom should have

formed a stronger bonding at BCC $\Sigma 3$ (112) GB than that at BCC PdCu bulk, which would therefore bring about higher hydrogen solubility at BCC $\Sigma 3$ (112) GB of PdCu shown in Fig. 2.

3.2. Hydrogen diffusivity at $\Sigma 3$ grain boundary of PdCu

It is well known that hydrogen diffusion in GB of various metal materials is quite different from that in single crystals or within the grain. Interestingly, some experimental studies reveal that hydrogen diffusion in GB of nickel is much faster than that in bulks due to a lower activation energy,^{3,6} whereas the hydrogen diffusivity in GB of aluminum and nickel with fine grains is smaller than that in the bulks.^{7,10,11,13} As to hydrogen diffusion

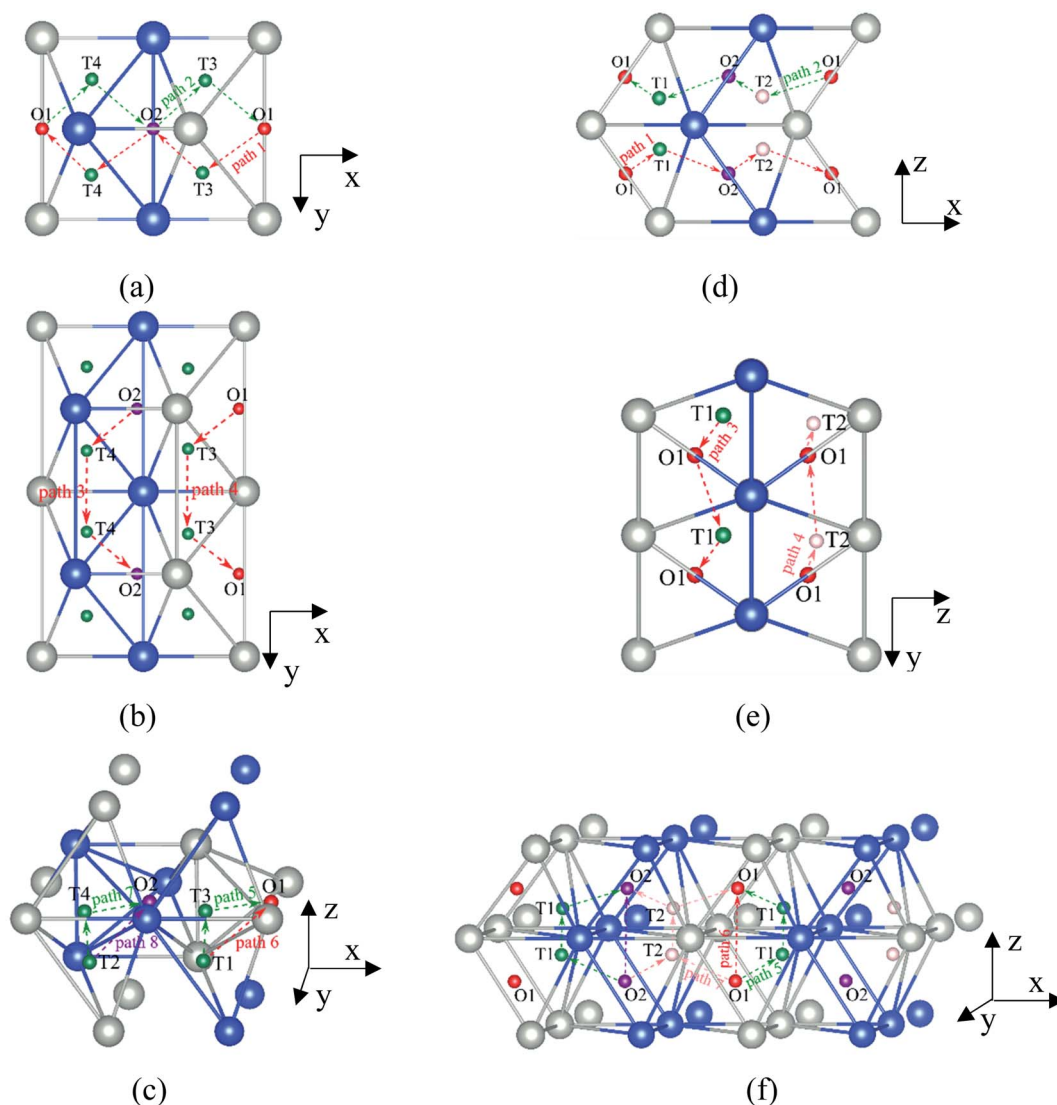


Fig. 4 The three layers atomic structure of the black dotted area in Fig. 1. And hydrogen diffusion through BCC PdCu $\Sigma 3$ (112) GB along (a) x direction (along GB) (path 1: O1 \rightarrow T3 \rightarrow O2 \rightarrow T4 \rightarrow O1, path 2: O2 \rightarrow T3 \rightarrow O1 \rightarrow T4 \rightarrow O2), (b) y direction (along GB) (path 3: O2 \rightarrow T4 \rightarrow T4 \rightarrow O2, path 4: O1 \rightarrow T3 \rightarrow T3 \rightarrow O1), and (c) z direction (across GB) (path 5–8: T1 \rightarrow T3 \rightarrow O1 \rightarrow T3 \rightarrow T1, T1 \rightarrow O1 \rightarrow T1, T2 \rightarrow T4 \rightarrow O2 \rightarrow T4 \rightarrow T2, and T2 \rightarrow O2 \rightarrow T2), respectively; hydrogen diffusion through FCC PdCu $\Sigma 3$ (111) GB along (d) x direction (along GB) (path 1: O1 \rightarrow T1 \rightarrow O2 \rightarrow T2 \rightarrow O1, path 2: O1 \rightarrow T2 \rightarrow O2 \rightarrow T1 \rightarrow O1), (e) y direction (along GB) (path 3: O1 \rightarrow T1 \rightarrow O1, path 4: O1 \rightarrow T2 \rightarrow O1), and (f) z direction (across GB) (path 5–7: O1 \rightarrow T1 \rightarrow T1 \rightarrow O1, O1 \rightarrow O1, and O1 \rightarrow T2 \rightarrow T2 \rightarrow O1). The x-axis is $[111]_{\text{BCC}}$ and $[011]_{\text{FCC}}$, the y-axis is $[110]_{\text{BCC}}$ and $[110]_{\text{FCC}}$, the z-axis is $[112]_{\text{BCC}}$ and $[111]_{\text{FCC}}$. The gray and blue spheres represent Pd and Cu atoms, respectively. The small balls stand for the interstitial sites.



in PdCu membranes, the reported experimental results are mainly concentrated on the polycrystalline phases,^{19–22,24–35,37,39} while the theoretical investigations are all related to single crystals.^{17,23,36,38,40–42} It is therefore of importance to find out the effect of GB on hydrogen diffusion of PdCu by means of first principles calculation.

According to lattice symmetry, the BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GBs of PdCu have eight and seven hydrogen diffusion paths, respectively, which are shown clearly in Fig. 4. For BCC $\Sigma 3$ (112) GB, the four hydrogen diffusion paths along GB are O1 \rightarrow T3 \rightarrow O2 \rightarrow T4 \rightarrow O1, O2 \rightarrow T3 \rightarrow O1 \rightarrow T4 \rightarrow O2, O2 \rightarrow T4 \rightarrow T4 \rightarrow O2, and O1 \rightarrow T3 \rightarrow T3 \rightarrow O1, while the four diffusion paths across GB are T1 \rightarrow T3 \rightarrow O1 \rightarrow T3 \rightarrow T1, T1 \rightarrow O1 \rightarrow T1, T2 \rightarrow T4 \rightarrow O2 \rightarrow T4 \rightarrow T2, and T2 \rightarrow O2 \rightarrow T2. Similarly, the hydrogen diffusion paths along GB of FCC $\Sigma 3$ (111) are O1 \rightarrow T1 \rightarrow O2 \rightarrow T2 \rightarrow O1, O1 \rightarrow T2 \rightarrow O2 \rightarrow T1 \rightarrow O1, O1 \rightarrow T1 \rightarrow O1, and O1 \rightarrow T2 \rightarrow O1, and the diffusion paths across GB are O1 \rightarrow T1 \rightarrow T1 \rightarrow O1, O1 \rightarrow O1, and O1 \rightarrow T2 \rightarrow T2 \rightarrow O1.

After a series of calculations, the derived energy barrier ($E_{d,i}$) of various diffusion paths are summarized in Table 2. It should be pointed out that the highest diffusion barrier has been regarded as the activation energy ($E_{a,i}$) for each diffusion path. Several features could be observed from Table 2. Firstly, for BCC $\Sigma 3$ (112) GB of PdCu, diffusion path 5 possesses the lowest activation energies of 0.141 eV among the four paths across GB, which is much smaller than the corresponding lowest $E_{a,i}$ of 0.243 eV of paths 1 and 3 along GB. This comparison suggests that hydrogen is energetically more favorable to diffuse across the BCC $\Sigma 3$ (112) GB of PdCu, rather than diffusion along GB. On the other hand, the lowest $E_{a,i}$ along and across FCC $\Sigma 3$ (111) GB of PdCu are the same value of 0.331 eV, indicating that hydrogen has an equal probability to diffuse along and across FCC $\Sigma 3$ (111) GB of PdCu.

Secondly, the lowest activation energy (0.141 eV) of hydrogen diffusion in BCC $\Sigma 3$ (112) GB of PdCu seems much bigger than the corresponding value of 0.057 eV of BCC PdCu bulk.³⁸ This characteristic implies that hydrogen diffusion in BCC $\Sigma 3$ (112) GB should be energetically more difficult than that in BCC PdCu bulk. On the contrary, the lowest activation energy (0.331 eV) of hydrogen diffusion in FCC $\Sigma 3$ (111) GB is slightly lower than the derived value of 0.347 eV of FCC PdCu bulk,³⁸ suggesting that it should be easier for hydrogen to diffuse in FCC $\Sigma 3$ (111) GB than in FCC PdCu bulk.

Thirdly, the lowest activation energies (0.243 and 0.141 eV) of hydrogen diffusion along and across BCC $\Sigma 3$ (112) GB of PdCu are much smaller than the corresponding value of 0.331 eV of FCC $\Sigma 3$ (111) GB of PdCu. A similar feature has been discovered about the relative magnitude of the lowest activation energies (0.057 and 0.347 eV) of BCC and FCC bulks of PdCu.³⁸ In other words, hydrogen has energetically easier diffusion in BCC $\Sigma 3$ (112) GB and bulks of PdCu than its FCC counterpart. Interestingly, the difference of lowest activation energies of hydrogen diffusion between BCC and FCC GB of PdCu becomes smaller that the corresponding value between BCC and FCC PdCu bulks. It should be pointed out that the ballistic diffusion may

Table 2 Diffusion barrier ($E_{d,i}$) of each hydrogen diffusion path in BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GBs of PdCu and the corresponding imaginary frequency at saddle point is listed in the parenthesis after $E_{d,i}$. The unit of imaginary frequency is THz. The corresponding $E_{a,i}$ of BCC and FCC PdCu bulks are 0.057 and 0.347 eV, respectively³⁸

Direction	Diffusion path	BCC $\Sigma 3$ (112)		FCC $\Sigma 3$ (111)	
		$E_{d,i}$ (eV)	Frequency (THz)	$E_{d,i}$ (eV)	Frequency (THz)
Along GB	1	0.088(13.71)	T3 \rightarrow T3	0.331(18.12)	T1 \rightarrow T1
	2	0.161(10.50)	T3 \rightarrow T3	0.401(26.90)	T2 \rightarrow T2
	3	0.243(13.02)	T4 \rightarrow T4	0.331(18.12)	T1 \rightarrow T1
	4	0.088(13.71)	T3 \rightarrow T3	0.401(26.90)	T2 \rightarrow T2
Across GB	5	0.112(11.36)	T3 \rightarrow T3	0.331(18.12)	T1 \rightarrow T1
	6	0.101(10.32)	O1 \rightarrow O1	0.382(33.57)	O1 \rightarrow O1
	7	0.193(15.32)	T4 \rightarrow T4	0.401(26.90)	T2 \rightarrow T2
	8	0.173(11.33)	O2 \rightarrow O2	0.401(26.90)	T2 \rightarrow T2
Along GB	1	0.195(10.50)	T3 \rightarrow T3	0.331(18.12)	T1 \rightarrow T1
	2	0.045(13.71)	T3 \rightarrow T3	0.401(26.90)	T2 \rightarrow T2
	3	0.096(15.11)	T4 \rightarrow T4	0.045(18.12)	O1 \rightarrow O1
	4	1.091(14.63)	T3 \rightarrow T3	0.121(26.90)	O1 \rightarrow O1
	5	0.045(13.71)	T3 \rightarrow T3	0.096(26.40)	T1 \rightarrow T1
	6	0.193(10.32)	O1 \rightarrow O1	0.045(18.12)	O1 \rightarrow O1
	7	0.068(13.02)	T4 \rightarrow T4	0.215(34.92)	T2 \rightarrow T2
	8	0.187(11.33)	O2 \rightarrow O2	0.121(26.90)	O1 \rightarrow O1
Across GB	1	0.243(13.02)	O2 \rightarrow O2	0.140(20.74)	O2 \rightarrow O2
	2	0.443(17.74)	O1 \rightarrow O1	0.055(14.61)	O2 \rightarrow O2
	3	0.068(13.02)	T4 \rightarrow T4	0.045(18.12)	O1 \rightarrow O1
	4	0.045(13.71)	T3 \rightarrow T3	0.121(26.90)	O1 \rightarrow O1
	5	0.088(13.71)	O1 \rightarrow O1	0.096(26.40)	T1 \rightarrow T1
	6	0.193(10.32)	T1 \rightarrow T1	0.045(18.12)	O1 \rightarrow O1
	7	0.068(13.02)	O2 \rightarrow O2	0.215(34.92)	T2 \rightarrow T2
	8	0.187(11.33)	O2 \rightarrow O2	0.121(26.90)	O1 \rightarrow O1



probably occur at the GBs and further studies are welcome to find out this effect on H diffusion at PdCu GB in the future.

We now turn to calculate the diffusivity of hydrogen (D) through the GB of PdCu by means of the following formula:⁵⁷

$$D = \sum p_i D_{0,i} \exp\left(\frac{-E_{a,i}}{k_B T}\right) \Theta, \quad (8)$$

$$p_s = \frac{C_i}{C}, \quad (9)$$

$$\Theta = 1 - \frac{C_i}{x_i}, \quad (10)$$

where C_i is the hydrogen solubility of initial interstitial site on path i ; x_i is the number of initial interstitial site; $E_{a,i}$ is the activation energy of the path i ; k_B is the Boltzmann constant. And the pre-factor $D_{0,i}$ could be calculated by:⁵⁵

$$D_{0,i} = \frac{1}{6} \lambda_i^2 \Gamma_i, \quad (11)$$

$$\Gamma_i = \frac{\prod_{m=1}^3 v_m f(hv_m/2k_B T)}{\prod_{n=1}^2 v_n f(hv_n/2k_B T)}, \quad (12)$$

$$f(x) = \sinh(x)/x, \quad (13)$$

where v_m and v_n are vibration frequencies at the initial and transition states, respectively; and λ_i is the jump distance.

The temperature-dependent diffusivity of hydrogen through BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GB of PdCu is thus obtained and shown in Fig. 5. One can discern from this figure that GB has a quite effect on hydrogen diffusivity of BCC and FCC PdCu. For BCC PdCu, the curve of hydrogen diffusivity of GB is well below that of bulk,³⁸ signifying that GB should decrease hydrogen diffusivity of BCC PdCu. Such a feature is consistent with

similar experimental observations regarding hydrogen diffusion in GB of nickel and aluminum.^{7,9,10,58}

On the contrary, hydrogen diffusivity of FCC $\Sigma 3$ (111) GB of PdCu at each temperature is bigger than that of FCC PdCu bulk.³⁸ It therefore follows that GB has an important effect to increase hydrogen diffusivity of FCC PdCu, which is just opposite to the decrease of hydrogen diffusivity of BCC PdCu as a result of GB. This increase (decrease) of hydrogen diffusivity due to GB is mainly attributed to the lower (higher) activation energies shown in Table 2. The higher hydrogen diffusivity of FCC $\Sigma 3$ (111) GB of PdCu from the present study is also compatible with similar experimental evidence of nickel in the literature.^{3,6,57,59} It should be noted that the calculated hydrogen diffusivities of BCC PdCu bulk³⁸ and FCC PdCu $\Sigma 3$ (111) GB agree well with the corresponding calculation values in the literature.²⁰

In addition, hydrogen diffusivity through BCC $\Sigma 3$ (112) GB of PdCu is significantly higher than that of FCC $\Sigma 3$ (111) GB, and the same statement could be made for hydrogen diffusivity at BCC and FCC bulks of PdCu.³⁸ It can be seen clearly from Fig. 5 that at each temperature, the descending sequence of hydrogen diffusivity of PdCu is as follows: BCC bulk \rightarrow BCC $\Sigma 3$ (112) GB \rightarrow FCC $\Sigma 3$ (111) GB \rightarrow FCC bulk. Such a comparison confirms that the BCC PdCu $\Sigma 3$ (112) GB and bulk have much higher hydrogen diffusivity than its FCC counterpart. It should be pointed out that the hydrogen defect complexes would possibly arise in the GB of PdCu and have some effects on H diffusivities and solubilities.⁶⁰

4. Conclusions

First principles calculations have been performed to comparatively study the effects of grain boundary on hydrogen solubility and hydrogen diffusivity at PdCu phases with BCC and FCC structures. It is found that the hydrogen solubility at BCC PdCu $\Sigma 3$ (112) GB is higher than that at BCC PdCu bulk. On the contrary, the hydrogen solubility at FCC PdCu $\Sigma 3$ (111) GB is lower than that at its bulk. Calculations also reveal that hydrogen diffusion across BCC PdCu $\Sigma 3$ (112) GB is easier than along the corresponding GB, while hydrogen diffusion along and across FCC PdCu $\Sigma 3$ (111) GB should have an equal probability. Moreover, BCC $\Sigma 3$ (112) GB could impede hydrogen diffusion and FCC PdCu $\Sigma 3$ (111) GB would accelerate hydrogen diffusion. The calculated results could provide a deep understanding of the effects of GB on hydrogen solubility and diffusivity of BCC and FCC PdCu.

Conflicts of interest

There are no conflicts to declare.

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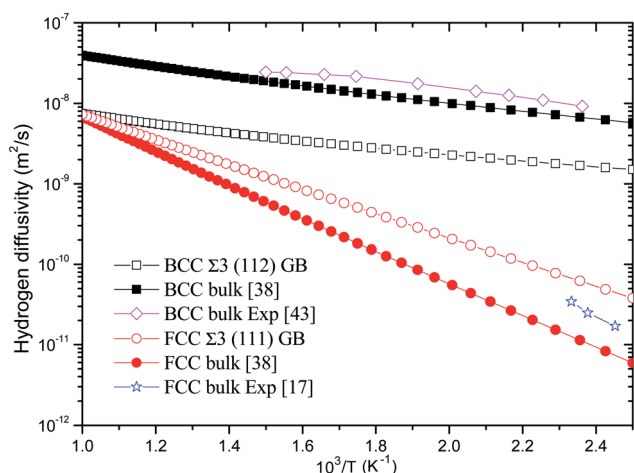


Fig. 5 Temperature-dependent hydrogen diffusivity through BCC $\Sigma 3$ (112) and FCC $\Sigma 3$ (111) GB of PdCu. The corresponding values of BCC and FCC PdCu bulks³⁸ are also listed for comparison.



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References

- 1 B. Ladna and H. K. Birnbaum, SIMS study of hydrogen at the surface and grain boundaries of nickel bicrystals, *Acta Metall.*, 1987, **35**, 2537–2542.
- 2 A. Kimura and H. K. Birnbaum, Hydrogen induced grain boundary fracture in high purity nickel and its alloys—enhanced hydrogen diffusion along grain boundaries, *Acta Metall.*, 1988, **36**, 757–766.
- 3 T. M. Harris and M. Latanision, Grain boundary diffusion of hydrogen in nickel, *Metall. Trans. A*, 1991, **22**, 351–355.
- 4 G. Palumbo, D. M. Doyle, A. M. Elsharik, U. Erb and K. T. Aust, Intercrystalline hydrogen transport in nanocrystalline nickel, *Scr. Metall. Mater.*, 1991, **25**, 679–684.
- 5 T. Tsuru and R. M. Latanision, Grain boundary transport of hydrogen in nickel, *Scr. Metall.*, 1982, **16**, 575–578.
- 6 J. Y. Lee and S. M. Lee, Hydrogen trapping phenomena in metals with B. C. C. and F. C. C. crystals structures by the desorption thermal analysis technique, *Surf. Coat. Technol.*, 1986, **28**, 301–314.
- 7 M. Ichimura, Y. Sasajima and M. Imabayashi, Grain Boundary Effect on Diffusion of Hydrogen in Pure Aluminum, *Mater. Trans., JIM*, 1991, **32**, 1109–1114.
- 8 U. Stuhr, T. Striffler, H. Wipf, H. Natter, B. Wettmann, S. Janssen, *et al.*, An investigation of hydrogen diffusion in nanocrystalline Pd by neutron spectroscopy, *J. Alloys Compd.*, 1997, **253**, 393–396.
- 9 R. M. Latanision and M. Kurkela, Hydrogen Permeability and Diffusivity in Nickel and Ni-Base Alloys, *Corrosion*, 1983, **39**, 174–181.
- 10 J. Yao and J. R. Cahoon, Experimental studies of grain boundary diffusion of hydrogen in metals, *Acta Metall. Mater.*, 1991, **39**, 119–126.
- 11 J. Yao and J. R. Cahoon, Theoretical modeling of grain boundary diffusion of hydrogen and its effect on permeation curves, *Acta Metall. Mater.*, 1991, **39**, 111–118.
- 12 A. Pedersen and H. Jonsson, Simulations of hydrogen diffusion at grain boundaries in aluminum, *Acta Mater.*, 2009, **57**, 4036–4045.
- 13 D. D. Stefano, M. Mrovec and C. Elsasser, First-principles investigation of hydrogen trapping and diffusion at grain boundaries in nickel, *Acta Mater.*, 2015, **98**, 306–312.
- 14 T. Mutschele and R. Kirchheim, Segregation and diffusion of hydrogen in grain boundaries of palladium, *Scr. Metall.*, 1987, **21**, 135–140.
- 15 J. A. Eastman, L. J. Thompson and B. J. Kestel, Narrowing of the palladium–hydrogen miscibility gap in nanocrystalline palladium, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **48**, 84–92.
- 16 U. Stuhr, H. Wipf, T. J. Udovic, J. Weismuller and H. Gleiter, Inelastic neutron scattering study of hydrogen in nanocrystalline Pd, *Nanostruct. Mater.*, 1995, **6**, 555–558.
- 17 S. M. Opalka, W. Huang, D. Wang, T. B. Flanagan, O. M. Lovvik, S. C. Emerson, *et al.*, Hydrogen interactions with the PdCu ordered B2 alloy, *J. Alloys Compd.*, 2007, **446**, 583–587.
- 18 L. Yuan, A. Goldbach and H. Xu, Permeation hysteresis in PdCu membranes, *J. Phys. Chem. B*, 2008, **112**, 12692–12695.
- 19 Y. Matsumura, A basic study of high-temperature hydrogen permeation through bcc Pd–Cu alloy foil, *Sep. Purif. Technol.*, 2014, **138**, 130–137.
- 20 N. A. Al-Mufachi, S. Nayeboossadri, J. D. Speight, W. Bujalski, R. Steinberger-Wilckens and D. Book, Effects of thin film Pd deposition on the hydrogen permeability of Pd₆₀Cu₄₀ wt% alloy membranes, *J. Membr. Sci.*, 2015, **493**, 580–588.
- 21 E. Acha, J. Requies, V. L. Barrio, J. F. Cambra, M. B. Guemez, P. L. Arias, *et al.*, PdCu membrane applied to hydrogen production from methane, *J. Membr. Sci.*, 2012, **415**, 66–74.
- 22 N. Almufachi and R. Steinbergerwilckens, X-ray diffraction study on the effects of hydrogen on Pd₆₀Cu₄₀ wt% foil membranes, *J. Membr. Sci.*, 2018, **545**, 266–274.
- 23 M. C. Gao, L. Ouyang and O. N. Dogan, First principles screening of B2 stabilizers in CuPd-based hydrogen separation membranes: (1) substitution for Pd, *J. Alloys Compd.*, 2013, **574**, 368–376.
- 24 N. Endo, Y. Furukawa, K. Goshome, S. Yaegashi, K. Mashiko and M. Tetsuhiko, Characterization of mechanical strength and hydrogen permeability of a PdCu alloy film prepared by one-step electroplating for hydrogen separation and membrane reactors, *Int. J. Hydrogen Energy*, 2019, **44**, 8290–8297.
- 25 H. T. Hoang, H. D. Tong, F. C. Gielens, H. Jansen and M. C. Elwenspoek, Fabrication and characterization of dual sputtered Pd–Cu alloy films for hydrogen separation membranes, *Mater. Lett.*, 2004, **58**, 525–528.
- 26 V. M. Ievlev, A. S. Prizhimov and A. I. Dontsov, Structure of the α – β interface in a PdCu solid solution, *Phys. Solid State*, 2020, **62**, 59–64.
- 27 S. Nayeboossadri, J. Speight and D. Book, Pd–Cu–M (M = Y, Ti, Zr, V, Nb, and Ni) Alloys for the Hydrogen Separation Membrane, *ACS Appl. Mater. Interfaces*, 2017, **9**, 2650–2661.
- 28 M. H. Martin, J. Galipaud, A. Tranchot, L. Roue and D. Guay, Measurements of hydrogen solubility in Cu_xPd_{100–x} thin films, *Electrochim. Acta*, 2013, **90**, 615–622.
- 29 A. Kulprathipanja, G. O. Alptekin, J. L. Falconer and J. D. Way, Pd and Pd–Cu membranes: inhibition of H₂ permeation by H₂S, *J. Membr. Sci.*, 2005, **254**, 49–62.
- 30 J. Tang, S. Yamamoto, T. Koitaya, A. Yoshigoe, T. Tokunaga, K. Mukai, *et al.*, Mass transport in the PdCu phase structures during hydrogen adsorption and absorption studied by XPS under hydrogen atmosphere, *Appl. Surf. Sci.*, 2019, **480**, 419–426.
- 31 H. Jia, P. Wu, G. Zeng, E. Salascolera, A. Serrano, G. R. Castro, *et al.*, High-temperature stability of Pd alloy membranes containing Cu and Au, *J. Membr. Sci.*, 2017, **544**, 151–160.
- 32 R. J. Westerwaal, E. A. Bouman, W. G. Haije, H. Schreuders, S. Dutta, M. Y. Wu, *et al.*, The hydrogen permeability of Pd–Cu based thin film membranes in relation to their structure: a combinatorial approach, *Int. J. Hydrogen Energy*, 2015, **40**, 3932–3943.



- 33 L. Yuan, A. Goldbach and H. Xu, Real-time monitoring of metal deposition and segregation phenomena during preparation of PdCu membranes, *J. Membr. Sci.*, 2008, **322**, 39–45.
- 34 A. Goldbach, L. Yuan and H. Xu, Impact of the fcc/bcc phase transition on the homogeneity and behavior of PdCu membranes, *Sep. Purif. Technol.*, 2010, **73**, 65–70.
- 35 L. Yuan, A. Goldbach and H. Xu, Segregation and H₂ transport rate control in body-centered cubic PdCu membranes, *J. Phys. Chem. B*, 2007, **111**, 10952–10958.
- 36 P. Kamakoti and D. S. Sholl, A comparison of hydrogen diffusivities in Pd and CuPd alloys using density functional theory, *J. Membr. Sci.*, 2003, **225**, 145–154.
- 37 S. Nayeboadri, J. Speight and D. Book, Effects of low Ag additions on the hydrogen permeability of Pd–Cu–Ag hydrogen separation membranes, *J. Membr. Sci.*, 2014, **451**, 216–225.
- 38 L. C. Liu, J. W. Wang, Y. H. He and H. R. Gong, Solubility, diffusivity, and permeability of hydrogen at PdCu phases, *J. Membr. Sci.*, 2017, **542**, 24–30.
- 39 C. Zhao, A. Goldbach and H. Xu, Low-temperature stability of body-centered cubic PdCu membranes, *J. Membr. Sci.*, 2017, **542**, 60–67.
- 40 L. C. Liu, H. R. Gong and S. F. Zhou, Hydrogen flux of BCC and FCC PdCuAg membranes, *Int. J. Hydrogen Energy*, 2019, **44**, 31160–31171.
- 41 C. Ling and D. S. Sholl, Using first-principles calculations to predict surface resistances to H₂ transport through metal alloy membranes, *J. Membr. Sci.*, 2007, **303**, 162–172.
- 42 L. C. Liu, H. R. Gong, S. F. Zhou and X. Gong, Adsorption, diffusion, and permeation of hydrogen at PdCu surfaces, *J. Membr. Sci.*, 2019, **588**, 117206.
- 43 C. Decaux, R. Ngameni, D. Solas, S. Grigoriev and P. Millet, Time and frequency domain analysis of hydrogen permeation across PdCu metallic membranes for hydrogen purification, *Int. J. Hydrogen Energy*, 2010, **35**, 4883–4892.
- 44 G. Kresse and J. Furthmüller, Efficiency of *ab initio* total energy calculations for metals and semiconductors using a plane-wave basis set, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 45 G. Kresse and J. Furthmüller, Efficient iterative schemes for *ab initio* total-energy calculations using a plane-wave basis set, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 46 L. Sun, H. R. Gong and X. Gong, Magnetic ground state of face-centered-cubic structure of iron, *J. Phys.: Condens. Matter*, 2020, **32**, 165806.
- 47 C. Y. Wu, L. Sun, C. Liang, H. R. Gong, M. L. Chang and D. C. Chen, Electronic structures and thermoelectric properties of polytype phases of bismuth, *J. Phys. Chem. Solids*, 2019, **134**, 52–57.
- 48 P. E. Blochl, Projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**, 17953–17979.
- 49 G. Kresse and D. P. Joubert, From ultrasoft pseudopotentials to the projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758–1775.
- 50 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 51 M. Methfessel and A. Paxton, High-precision sampling for Brillouin-zone integration in metals, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1989, **40**, 3616–3621.
- 52 P. E. Blochl, O. Jepsen and O. K. Andersen, Improved tetrahedron method for Brillouin-zone integrations, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **49**, 16223–16233.
- 53 J. Xu, Y. Jiang, L. Yang and J. Li, Assessment of the CSL and SU models for bcc-Fe grain boundaries from first principles, *Comput. Mater. Sci.*, 2016, **122**, 22–29.
- 54 G. Henkelman and H. Jónsson, Improved tangent estimate in the nudged elastic band method for finding minimum energy paths and saddle points, *J. Chem. Phys.*, 2000, **113**, 9978–9985.
- 55 L. Qin and C. Jiang, First-principles based modeling of hydrogen permeation through Pd–Cu alloys, *Int. J. Hydrogen Energy*, 2012, **37**, 12760–12764.
- 56 Y. Du, L. Ismer, J. Rogal, T. Hickel, J. Neugebauer and R. Drautz, First-principles study on the interaction of H interstitials with grain boundaries in α - and γ -Fe, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 144121.
- 57 X. Zhou, N. Mousseau and J. Song, Is Hydrogen Diffusion along Grain Boundaries Fast or Slow? Atomistic Origin and Mechanistic Modeling, *Phys. Rev. Lett.*, 2019, **122**, 215501.
- 58 J. Wang, N. Li and A. Misra, Structure and stability of $\Sigma 3$ grain boundaries in face centered cubic metals, *Philos. Mag.*, 2013, **93**, 315–327.
- 59 J. Yang, J. Huang, Z. Ye, D. Fan, S. Chen and Y. Zhao, First-principles investigation on the interaction of Boron atom with nickel part II: absorption and diffusion at grain boundary, *J. Alloys Compd.*, 2017, **708**, 1089–1095.
- 60 J. Polfus, O. Løvvik, R. Bredesen and T. A. Peters, Hydrogen Induced Vacancy Clustering and Void Formation Mechanisms at Grain Boundaries in Palladium, *Acta Mater.*, 2020, **195**, 708–719.

