


 Cite this: *RSC Adv.*, 2021, **11**, 2866

Water-assisted synthesis of highly stable CsPbX₃ perovskite quantum dots embedded in zeolite-Y†

 Hui Cheng,^{ab} Yanfeng Yin,^b Jianbo Tang,^b Donghua Fan,^{*a} Jan J. Huang^{*a} and Shengye Jin ^{*b}

All-inorganic perovskite materials have emerged as highly promising materials for solar cells and photoelectronic applications. However, the poor stability of perovskites in ambient conditions significantly hampers their practical applications. In this work, we report a three-step synthesis of size tunable CsPbX₃ (X = Br, Cl, or I) quantum dots (QDs) embedded in zeolite-Y (CsPbX₃-Y), which involves efficient chemical transformation of non-luminescent Cs₄PbX₆ to highly luminescent CsPbX₃ by stripping CsX through an interfacial reaction with water. We show that the size and the emission of CsPbX₃ in CsPbX₃-Y can be tuned by the amount of water added as well as the halide composition. More importantly, the as-prepared CsPbX₃-Y show significantly enhanced stability against moisture upon protection by zeolite-Y. This work not only reports a new pathway for the preparation of highly luminescent CsPbX₃ but also provided new insights into the chemical transformation behavior and stabilization mechanism of these emerging perovskites.

Received 29th September 2020

Accepted 25th November 2020

DOI: 10.1039/d0ra08311a

rsc.li/rsc-advances

Introduction

Benefiting from their compositional tunability and outstanding charge transport properties, all inorganic perovskites and hybrid lead-halide perovskites have attracted significant attention in recent years in the materials research community,^{1–4} particularly in solar cells,^{5–7} solid-state light-emitting diodes,^{8,9} photodetectors and lasers.^{10,11} However, the sensitivity of perovskites to moisture and their apparent instability upon contact with water significantly impede their practical applications.^{12–14} Recent work focusing on investigating the interaction mechanism between perovskites and water uncovered the existence of weak interactions between the organic and inorganic constituents,^{15,16} particularly at the surface of the perovskite, which is believed to be responsible for the rapid decomposition of perovskite upon exposure to H₂O.^{17–19} As a result, efforts have been devoted to modify the surface of perovskite in attempt to improve their stability against moisture.^{20–24} For example, organic ligands or polymers such as tri-*n*-octylphosphine,²⁵ polyhedral oligomeric silsesquioxane and titanium butoxide have been used to modify the surface to enhance the stability of perovskite quantum dots

(QDs).^{26,27} In addition to organic components,²⁸ many inorganic materials which exhibit good thermal resistance such as SiO₂, TiO₂ and Al₂O₃ have been used to protect perovskite QDs.^{4,27,29–32} More recently, porous zeolites have drawn increasing attention as hosts to encapsulate and stabilize perovskite QDs.³³ Zeolites are porous aluminosilicates, in which an open framework structure of [AlO₄]^{5–} and [SiO₄]^{4–} tetrahedra forms a regular 3D structure of cages and channels. As a result, they can accommodate a variety of charge balancing cations that are readily exchangeable and have been demonstrated to improve and stabilize emission from various types of luminescent centers that were introduced into the porous zeolite structure by ion exchange.³⁴

In this work, we report a successful three-step synthesis method for perovskite QDs embedded in zeolite-Y. The zeolite-Y is first exchanged with Cs⁺ ions by an ion-exchange reaction from aqueous solution. Cs⁺ ion in zeolite then reacts with Pb²⁺ and X[–] (with X = Cl, Br, or I) ions to form non-luminescent Cs₄PbX₆. In the final step, Cs₄PbX₆ is converted into CsPbX₃ by stripping CsX through an interfacial reaction with water. We not only show that the size and the emission property of CsPbX₃ in CsPbX₃-Y can be controlled by the amount of water absorbed and the halide composition but also demonstrate that the stability of CsPbX₃-Y is significantly enhanced against moisture.

Experimental section

Materials

Caesium bromide (CsBr, Alfa Aesar, 99.999%), oleic acid (OA, Sigma-Aldrich, 90%), ODE (Sigma-Aldrich, 90%), oleylamine (OAm, Aladdin, 80–90%), lead chloride (PbCl₂, Alfa Aesar,

^aSchool of Applied Physics and Materials, Wuyi University, Jiangmen, 529020, China

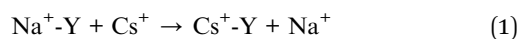
^bNational Key Laboratory of Molecular Reaction Dynamics, Dalian Institute of Chemical Physics, Chinese Academy of Sciences, Dalian, China, 116023

 † Electronic supplementary information (ESI) available: XRD of the CsPb(Cl_{0.4}Br_{0.6})₃-Y and CsPb(Br_{0.6}I_{0.4})₃-Y, the emission spectra and decay curves of different size CsPbBr₃-Y, the emission decay curves and lifetime of CsPbBr₃ QDs and CsPbBr₃-Y in environment and 100% RH, respectively, the emission spectra of CsPbBr₃-Y immersion in water, the images of Cs₄PbBr₆-Y and CsPbBr₃-Y in sun light and under UV. See DOI: 10.1039/d0ra08311a


99.999%), lead bromide (PbBr₂, Alfa Aesar, 99.999%), lead iodide (PbI₂, Alfa Aesar, 99.999%), *n*-trioctylphosphine (TOP, Strem, 97%), *n*-hexane (Aladdin, AR, 97%), isopropanol (AR, Fuyu Chemicals), Na⁺-zeolite-Y (SiO₂/Al₂O₃ = 5 : 1, Alfa Aesar). All solvents and chemicals were purchased from a commercial company and directly used without further purification.

Preparation of Cs⁺-Y

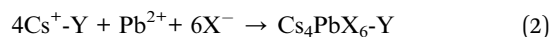
1.0 g of zeolite-Y and 0.01 mol CsBr were dispersed in 10 mL aqueous solution in single neck flask. The mixture was stirred in a water bath at 60 °C for 24 h. The products were centrifuged and washed with deionized water, and then dried in air at 80 °C for 12 h. The formation of Cs⁺-Y can be represented by the following chemical equation (eqn (1)).



Synthesis of Cs₄PbX₆-Y

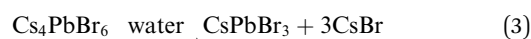
PbX₂ solution was prepared by mixing 0.188 mmol of PbX₂, 1.0 mL of OA, 1.0 mL of OAm in a 50 mL three-necked bottle along with 5.0 mL ODE. The mixture was dried for 30 min at 120 °C under vacuum with stirring. After that, the mixture was purged under N₂ flow until PbX₂ was completely dissolved. For preparing Cs₄PbCl₆-Y, PbCl₂ is used. PbCl₂ was dissolved at 150 °C with TOP (1 mL). The mixed PbX₂ with PbCl₂ : PbBr₂ = 0.4 : 0.6 and PbI₂ : PbBr₂ = 0.4 : 0.6 were used to prepare Cs₄Pb(Br_{0.6}Cl_{0.4})₆-Y and Cs₄Pb(Br_{0.6}I_{0.4})₆, respectively. Then Cs⁺-Y (0.5 g) was loaded into a 50 mL three-necked bottle along with 5.0 mL ODE. The mixture was dried for 30 min under vacuum at 120 °C. The reaction temperature was raised to 150 °C under N₂ atmosphere and maintained for another 30 min. The as-prepared PbX₂ solution was swiftly injected into the hot reaction mixture. The mixture was stirred about 10 min. For the larger halide ion I⁻, a longer reaction time (~15 min) was needed for PbX₂ to diffuse into the zeolite and react with the previously exchanged Cs⁺. The mixture was then cooled down using an ice-water bath. Finally, the obtained product was

washed with *n*-hexane and isopropanol twice followed by centrifugation. The Cs₄PbX₆ composites were dried at 80 °C for 5 h. The chemical equation corresponding to this reaction can be represented by (eqn (2)):



Synthesis of CsPbX₃-Y

A small amount of water was added into the Cs₄PbX₆-Y composites, which results into the conversion of Cs₄PbX₆-Y to CsPbX₃-Y (eqn (3)). The size of CsPbX₃ in CsPbX₃-Y can be controlled by the amount of water added.



Synthesis of CsPbBr₃ quantum dots (QDs)

In a typical synthesis of CsPbBr₃ QDs, the Cs-precursor was prepared according to the literature protocol.³⁵ Briefly, 0.188 mmol of PbBr₂, 1.0 mL of OA, and 1.0 mL of OAm were loaded into a 50 mL three-necked bottle along with 5.0 mL ODE. The mixture was dried for 30 min at 120 °C under vacuum and stirring. Then the reaction temperature was raised to 150 °C under N₂ atmosphere and maintained for another 30 min. The pre-prepared Cs-precursor solution (0.4 mL) was swiftly injected into the hot reaction mixture. The system was cooled down with an ice-water bath after reaction about 5 seconds. Finally, the obtained product was dispersed in *n*-hexane after centrifugation.

Characterization and general procedure

The powder X-ray diffraction (PXRD) patterns were collected using Empyrean-100 diffractometer with Cu Kα radiation (λ = 1.5406 Å) at 293 K. The data were collected at 5°–40° range and scan speed is 5.0 degree per min. A Bruker TENSOR27

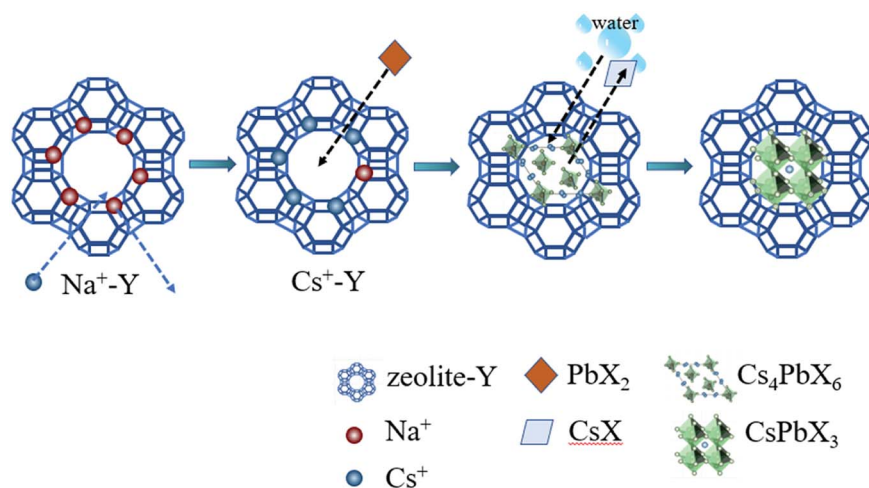


Fig. 1 Schematic illustration of the three-step synthesis of CsPbBr₃-Y composites.



spectrometer was applied to measure FT-IR (Fourier transform infrared) spectra (as KBr pressed pellets) in the 4000–400 cm^{-1} range. The surface morphology of the powder was observed using a JSM-7800F scanning electron microscope (SEM) with an accelerating voltage of 1.0 kV. Diffuse reflectance UV-visible spectra were obtained using Agilent Technologies Carry 60 UV-vis spectrophotometer. For the luminescence measurements, a laser source at 405 nm was employed. The luminescence signals were collected and directed into a fiber integrated spectrograph with spectra resolution < 1.0 nm. The fluorescence lifetime measurement setup used in this study was based on the time-correlated single photon counting technology (TCSPC). The excitation beam was picosecond pulse diode laser with 405 nm output wavelength. The optical detector was single photon counting module.

Results and discussion

Fig. 1 schematically illustrated the three-step synthesis of $\text{CsPbBr}_3\text{-Y}$. The first step is the exchange of Na^+ in zeolite-Y ($\text{Na}^+\text{-Y}$) by Cs^+ to form $\text{Cs}^+\text{-Y}$ (see details of procedure in Experimental section).³³ In the second step, PbX_2 was added into the suspension of $\text{Cs}^+\text{-Y}$ in ODE to form non-luminescent $\text{Cs}_4\text{PbX}_6\text{-Y}$. In the last step, non-luminescent $\text{Cs}_4\text{PbX}_6\text{-Y}$ is converted to luminescent $\text{CsPbX}_3\text{-Y}$ upon the addition of water. The formation of CsPbX_3 is believed to undergo stripping CsX from Cs_4PbX_6 via interfacial reaction with water.³⁶ The products formed during each step in Fig. 1 were confirmed by powder X-ray diffraction (XRD), Fourier transform infrared (FTIR) spectroscopy, and scanning electron microscope (SEM).

Fig. 2a shows the X-ray diffraction (XRD) patterns of the zeolite-Y, $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$. While the majority

of the diffraction angles of $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$ match that of zeolite-Y, the relative intensity of several diffraction peaks change in $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$ with respect to that in zeolite-Y. For example, compared to zeolite-Y, the intensity of the peaks at 12.41°, 14.38°, 18.66°, and 21.61° in $\text{Cs}^+\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$ increases, while the intensity of peaks at 15.78° and 20.27° decreases significantly. The similar results have been observed in previous literatures,³³ which can be attributed to the change of the unit cell of zeolite-Y while zeolite crystal structure remains fixed. These results suggest that the variations during ion-exchange reaction occur inside the pore of zeolite-Y, *i.e.* the formation of QDs occurs in mesopores of zeolite-Y although we cannot exclude that some QDs form on the surface of zeolite-Y. No XRD peaks originating from the CsPbBr_3 perovskite crystal structure are observed, which is likely due to the much smaller volume of embedded QDs (3–10 nm) than zeolite-Y (2 μm), resulting in much weaker diffraction signals in the former than the latter.

Fig. 2b compares the FTIR spectra of Zeolite-Y, $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$. The peak at 1642 cm^{-1} corresponds to the C=O symmetric stretching vibration of the deprotonated carboxylate group and the broad band at 3300 cm^{-1} can be assigned to the OH stretching vibration of the water molecule. The strong peaks at 2862 and 2937 cm^{-1} can be assigned to C-H bending that results from the hydrocarbon chains in organic species. The bands at 1140 cm^{-1} and 1021 cm^{-1} can be attributed to the vibration of Si-O-Si and Si-O-C, respectively.³⁷ Fig. 2c–f show scanning electron microscopy (SEM) images of zeolite-Y, $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$ and $\text{CsPbBr}_3\text{-Y}$ composites, respectively. The $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$, and $\text{CsPbBr}_3\text{-Y}$ all show similar morphology as that of zeolite-Y, suggesting the retain of

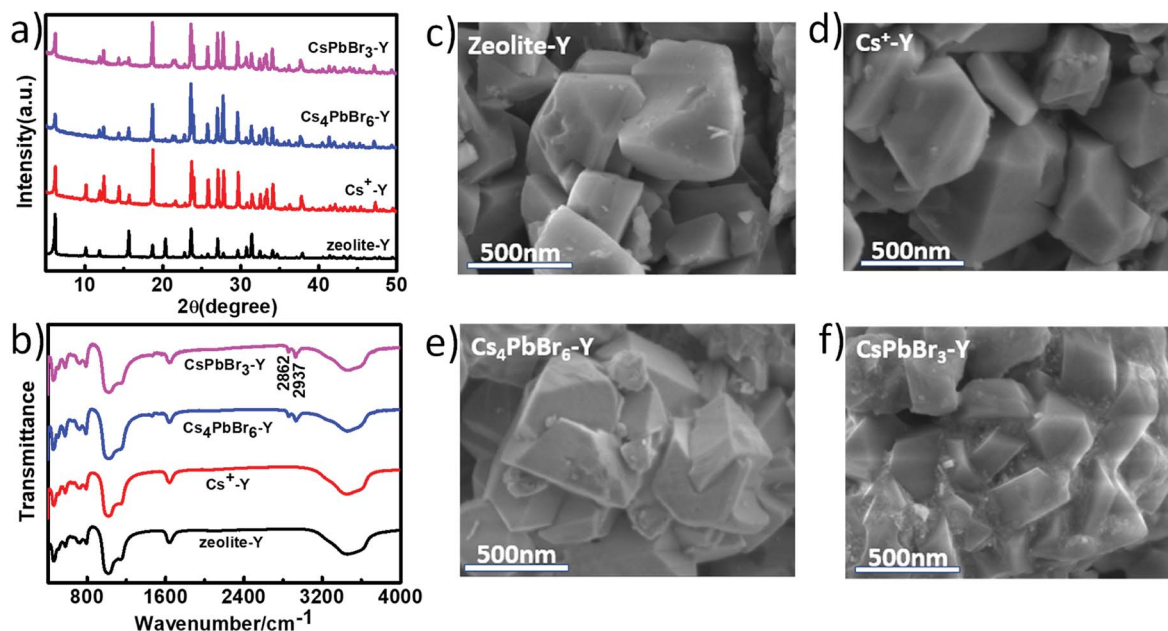


Fig. 2 PXRD patterns (a) and FTIR spectra (b) of zeolite-Y, $\text{Cs}^+\text{-Y}$, $\text{Cs}_4\text{PbBr}_6\text{-Y}$, $\text{CsPbBr}_3\text{-Y}$; the SEM images of Zeolite-Y (c), $\text{Cs}^+\text{-Y}$ (d), $\text{Cs}_4\text{PbBr}_6\text{-Y}$ (e), and $\text{CsPbBr}_3\text{-Y}$ (f).



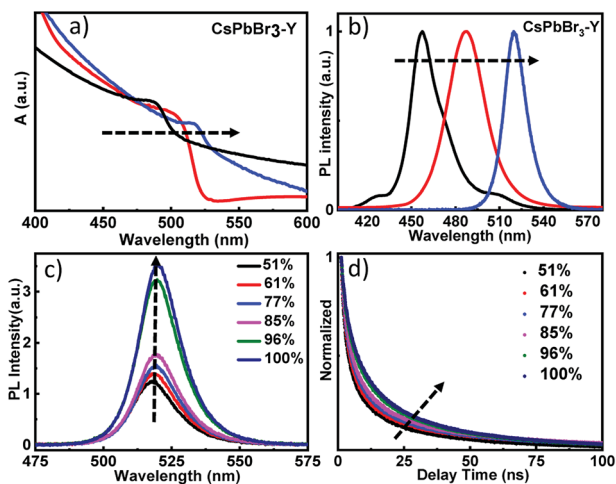


Fig. 3 The diffuse reflectance spectra (a) and the corresponding emission spectra (b) of CsPbBr₃-Y with different sizes of CsPbBr₃; (c) the emission spectra of CsPbBr₃-Y (520 nm) in different relative humidity; (d) the emission decay curves of CsPbBr₃-Y (520 nm) in different relative humidity.

zeolite structure among these samples, which is consistent with XRD results.

Fig. 3a shows the diffuse reflectance UV-visible spectra of as prepared CsPbBr₃-Y. Immediately after the formation of CsPbBr₃-Y, the first exciton peak of CsPbBr₃ QDs is centered at 470 nm (black plot). With increasing time of exposure to water, the first exciton peak of CsPbBr₃-Y shows a bathochromic shift, suggesting the gradual increase of QD size in CsPbBr₃-Y.^{29,38,39} The spectrum stops shifting when exciton peak reaches 520 nm (blue plot), which suggests that the QDs stop growing. These results are further supported from the steady state emission spectrum, where the emission peak for the above CsPbBr₃-Y samples also show red shift, ranging from 460 nm to 520 nm (Fig. 3b) upon exposure to water and stops shifting afterwards. These results are consistent with the diffuse reflectance UV-visible data above and suggest the growth of QDs in CsPbBr₃-Y with the amount of water absorbed. The sample when the size stops growing (*i.e.* the exciton peak at 520 nm) was used for further study and is denoted as CsPbBr₃-Y (520 nm).

The impact of relative humidity on the emission properties of CsPbBr₃-Y (520 nm) was examined under different humidity. As shown in Fig. 3c, the emission intensity of CsPbBr₃-Y (520 nm) gradually increases with increasing humidity.³⁶ Furthermore, the emission lifetime of CsPbBr₃-Y (520 nm) measured by time-resolved emission spectroscopy using time correlated single photon counting (TCSPC) technique becomes longer with increasing humidity (Fig. 3d). These results, consistent with the description in the previous literature,⁴⁰ can be attributed to that the nonradiative recombination has been inhibited with increasing humidity, consistent with the steady state emission results above.

In addition to the relative humidity, we also evaluated the stability of CsPbBr₃-Y (520 nm) by measuring the impact of environment on its emission intensity. As shown in Fig. 4a, the

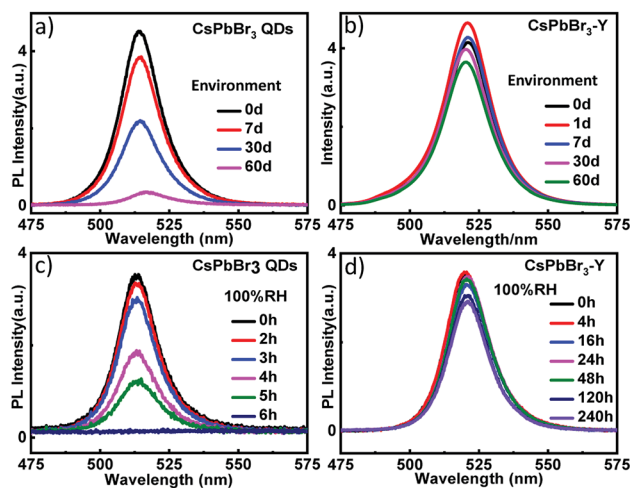


Fig. 4 The emission spectra of CsPbBr₃ QDs (a) and CsPbBr₃-Y (520 nm) (b) upon exposure to environment; the emission spectra of CsPbBr₃ QDs (c) and CsPbBr₃-Y (520 nm) (d) under 100% relative humidity.

emission intensity of CsPbBr₃ QDs with similar size without zeolite-Y decreases more than 51.3% after 30 days and the emission is almost completely quenched after 60 days. In contrast, the emission intensity in CsPbBr₃-Y (520 nm) only decays 15.4% by 30 days and 21.8% after 60 days (Fig. 4b). These results together suggest that embedding CsPbBr₃ in zeolite can significantly increase its stability in environment. The stability of CsPbBr₃-Y (520 nm) was further evaluated by exposing CsPbBr₃-Y (520 nm) to 100% relative humidity, where the intensity of CsPbBr₃ QDs is completely quenched after 6 hours (Fig. 4c), whereas the emission of CsPbBr₃-Y (520 nm) only decrease by 18.4% after 240 hours (Fig. 4d). Moreover, the emission lifetime of CsPbBr₃-Y (520 nm) is much longer than CsPbBr₃ QDs (Fig. S1a, S1b, Tables S1 and S2†). Based on these results, we can conclude that the stability of CsPbBr₃ in CsPbBr₃-Y composites is significantly improved.

It has been shown previously that the luminescence properties of CsPbX₃ are strongly dependent on their halide composition. In this work, we also examined the impact of halide composition to the emission property of CsPbX₃-Y. We first incorporated X (X = Cl or I) to Cs₄PbBr₆-Y with X : Br = 0.4 : 0.6, which forms Cs₄Pb(Br_{0.6}Cl_{0.4})₆-Y and Cs₄Pb(Br_{0.6}I_{0.4})₆-Y, respectively. After exposing Cs₄Pb(Br_{0.6}Cl_{0.4})₆-Y and Cs₄Pb(Br_{0.6}I_{0.4})₆-Y to water, the corresponding CsPb(Br_{0.6}Cl_{0.4})₃-Y and CsPb(Br_{0.6}I_{0.4})₃-Y were formed. Both Cs₄Pb(Br_{0.6}Cl_{0.4})₆-Y (i and iii in Fig. 5a) and Cs₄Pb(Br_{0.6}I_{0.4})₆-Y (i and iii in Fig. 5b) do not emit under sun light and ultraviolet light. After they absorb water, the color of CsPb(Br_{0.6}Cl_{0.4})₃-Y changes from white to yellow under sun light (ii in Fig. 5a) and cyan under ultraviolet (iv in Fig. 5a), while the color of CsPb(Br_{0.6}I_{0.4})₃-Y, change from white to red under sunlight (ii in Fig. 5b) and yellow under ultraviolet (iv in Fig. 5b). These results suggest that non-luminescent Cs₄Pb(Br_{0.6}X_{0.4})₆-Y (X = Cl or I) turn into luminescent CsPb(Br_{0.6}X_{0.4})₃-Y. It is worth mentioning that the reaction rate of the transformation process depends on the



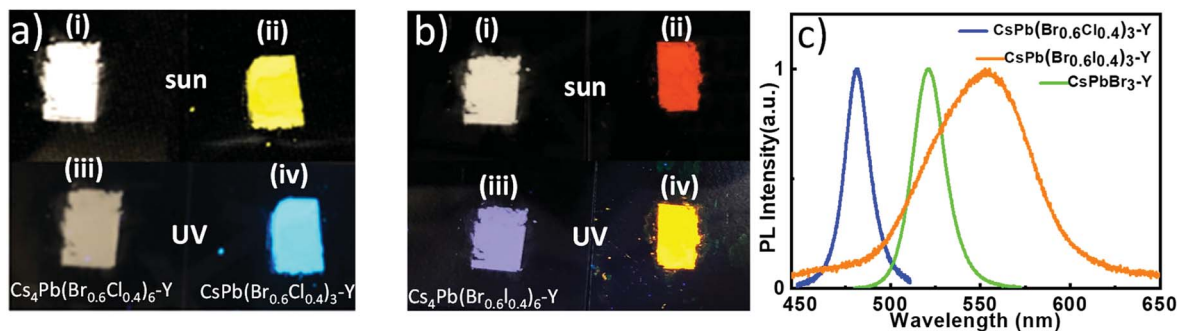


Fig. 5 (a) $\text{Cs}_4\text{Pb}(\text{Br}_{0.6}\text{Cl}_{0.4})_6\text{-Y}$ under sun light (i), $\text{CsPb}(\text{Br}_{0.6}\text{Cl}_{0.4})_3$ under sun light (ii), $\text{Cs}_4\text{Pb}(\text{Br}_{0.6}\text{Cl}_{0.4})_6\text{-Y}$ under UV (iii), $\text{CsPb}(\text{Br}_{0.6}\text{Cl}_{0.4})_3$ under UV (iv); (b) $\text{Cs}_4\text{Pb}(\text{Br}_{0.6}\text{I}_{0.4})_6\text{-Y}$ under sun light (i), $\text{CsPb}(\text{Br}_{0.6}\text{I}_{0.4})_3$ under sun light (ii), $\text{Cs}_4\text{Pb}(\text{Br}_{0.6}\text{I}_{0.4})_6\text{-Y}$ under UV (iii), $\text{CsPb}(\text{Br}_{0.6}\text{I}_{0.4})_3$ under UV (iv); (c) the emission spectra of $\text{CsPb}(\text{Br}_{0.6}\text{Cl}_{0.4})_3\text{-Y}$, $\text{CsPbBr}_3\text{-Y}$ and $\text{CsPb}(\text{Br}_{0.6}\text{I}_{0.4})_3\text{-Y}$.

halide composition. The transformation process of $\text{Cs}_4\text{-Pb}(\text{Br}_{0.6}\text{Cl}_{0.4})_6\text{-Y}$ to $\text{CsPb}(\text{Br}_{0.6}\text{Cl}_{0.4})_3\text{-Y}$ started immediately upon the addition of water, while it took several minutes to gradually see the color change for $\text{Cs}_4\text{Pb}(\text{Br}_{0.6}\text{I}_{0.4})_6\text{-Y}$. These phenomena can be attributed to the higher solubility of CsCl than that of CsI in water, resulting into the faster CsX -stripping process in the former than the latter. As shown in Fig. 5c, the emission spectra of these perovskite samples shifts significantly, where $\text{CsPb}(\text{Br}_{0.6}\text{Cl}_{0.4})_3\text{-Y}$ displays an emission peak at 480 nm, $\text{CsPb}(\text{Br}_{0.6}\text{I}_{0.4})_3\text{-Y}$ shows a relatively broad emission spectrum centered at 560 nm, both of which distinct from that of $\text{CsPbBr}_3\text{-Y}$, suggesting that the emission of $\text{CsPbX}_3\text{-Y}$ can be tuned by controlling the halide component.

Conclusions

In this work, we report a successful three-step method to synthesize perovskite QDs embedded in zeolite-Y, through which the efficient chemical transformation of non-luminescent Cs_4PbX_6 to highly luminescent CsPbX_3 occurs by stripping CsX from the former through an interfacial reaction with water. We show that the size and the emission of CsPbX_3 in $\text{CsPbX}_3\text{-Y}$ can be tuned by the amount of water added as well as the halide composition. More importantly, the as-prepared $\text{CsPbX}_3\text{-Y}$ show significantly enhanced stability against moisture upon protection by zeolite-Y. This work not only reports a new pathway for the preparation of highly luminescent CsPbX_3 but also provides important insights into the chemical transformation behavior and stabilization mechanism of these emerging perovskites.

Conflicts of interest

The authors declare no competing financial interests.

Acknowledgements

This work was supported by the Natural Science Foundation of Guangdong Province, 2018A030313460, Science and Technology Projects of Jiangmen (No. (2017) 307, (2017) 149, (2018)

352), Key Laboratory of Optoelectronic materials and Applications in Guangdong Higher Education (No: 2017KSYS011).

Notes and references

- 1 L. N. Quan, B. P. Rand, R. H. Friend, S. G. Mhaisalkar, T. W. Lee and E. H. Sargent, *Chem. Rev.*, 2019, **119**, 7444–7477.
- 2 S. Lee, J. H. Park, Y. S. Nam, B. R. Lee, B. Zhao, D. Di Nuzzo, E. D. Jung, H. Jeon, J. Y. Kim, H. Y. Jeong, R. H. Friend and M. H. Song, *ACS Nano*, 2018, **12**, 3417–3423.
- 3 W. Chen, H. Liu, R. Fan, P. Wang, T. Sun and Y. Yang, *ACS Appl. Mater. Interfaces*, 2020, **12**, 9851–9857.
- 4 L. Polavarapu, B. Nickel, J. Feldmann and A. S. Urban, *Adv. Energy Mater.*, 2017, **7**, 1700267.
- 5 X. Wang, L. Wang, T. Shan, S. Leng, H. Zhong, Q. Bao, Z. H. Lu, L. L. Deng and C. C. Chen, *Nano-Micro Lett.*, 2020, **12**, 84.
- 6 N. J. Jeon, H. Na, E. H. Jung, T. Y. Yang, Y. G. Lee, G. Kim, H. W. Shin, S. Il Seok, J. Lee and J. Seo, *Nat. Energy*, 2018, **3**, 682–689.
- 7 M. Anaya, G. Lozano, M. E. Calvo and H. Míguez, *Joule*, 2017, **1**, 769–793.
- 8 H. Cho, S. Jeong, M. Park, Y. Kim, C. Wolf, C. Lee and T. Lee, *Science*, 2015, **350**, 1222–1225.
- 9 Z. K. Tan, R. S. Moggaddam, M. L. Lai, P. Docampo, R. Higler, F. Deschler, M. Price, A. Sadhanala, L. M. Pazos, D. Credgington, F. Hanusch, T. Bein, H. J. Snaith and R. H. Friend, *Nat. Nanotechnol.*, 2014, **9**, 687–692.
- 10 X. He, Y. Qiu and S. Yang, *Adv. Mater.*, 2017, **29**, 201700775.
- 11 G. Xing, N. Mathews, S. S. Lim, N. Yantara, X. Liu, D. Sabba, M. Gratzel, S. Mhaisalkar and T. C. Sum, *Nat. Mater.*, 2014, **13**, 476–480.
- 12 Y. Wei, X. Deng, Z. Xie, X. Cai, S. Liang, P. a. Ma, Z. Hou, Z. Cheng and J. Lin, *Adv. Funct. Mater.*, 2017, **27**, 1703535.
- 13 C. Sun, Y. Zhang, C. Ruan, C. Yin, X. Wang, Y. Wang and W. W. Yu, *Adv. Mater.*, 2016, **28**, 10088–10094.
- 14 J. Hai, H. Li, Y. Zhao, F. Chen, Y. Peng and B. Wang, *Chem. Commun.*, 2017, **53**, 5400–5403.



- 15 K. K. Liu, Q. Liu, D. W. Yang, Y. C. Liang, L. Z. Sui, J. Y. Wei, G. W. Xue, W. B. Zhao, X. Y. Wu, L. Dong and C. X. Shan, *Light: Sci. Appl.*, 2020, **9**, 44.
- 16 C. Wu, Y. Zou, T. Wu, M. Ban, V. Pecunia, Y. Han, Q. Liu, T. Song, S. Duhm and B. Sun, *Adv. Funct. Mater.*, 2017, **27**, 1700338.
- 17 G. Niu, X. Guo and L. Wang, *J. Mater. Chem. A*, 2015, **3**, 8970–8980.
- 18 D. Zhang, Y. Xu, Q. Liu and Z. Xia, *Inorg. Chem.*, 2018, **57**, 4613–4619.
- 19 J. Ren, T. Li, X. Zhou, X. Dong, A. V. Shorokhov, M. B. Semenov, V. D. Krevchik and Y. Wang, *Chem. Eng. J.*, 2019, **358**, 30–39.
- 20 V. Malgras, J. Henzie, T. Takei and Y. Yamauchi, *Chem. Commun.*, 2017, **53**, 2359–2362.
- 21 Z. C. Kong, J. F. Liao, Y. J. Dong, Y. F. Xu, H. Y. Chen, D. B. Kuang and C. Y. Su, *ACS Energy Lett.*, 2018, **3**, 2656–2662.
- 22 Y. Wang, T. Wu, J. Barbaud, *et al.*, *Science*, 2019, **365**, 687–691.
- 23 M. Abdi-Jalebi, Z. Andaji-Garmaroudi, S. Cacovich, C. Stavrakas, B. Philippe, J. M. Richter, M. Alsari, E. P. Booker, E. M. Hutter, A. J. Pearson, S. Lilliu, T. J. Savenije, H. Rensmo, G. Divitini, C. Ducati, R. H. Friend and S. D. Stranks, *Nature*, 2018, **555**, 497–501.
- 24 L. Yang, T. Wang, Q. Min, B. Liu, Z. Liu, X. Fan, J. Qiu, X. Xu, J. Yu and X. Yu, *ACS Omega*, 2019, **4**, 6084–6091.
- 25 C. Lu, H. Li, K. Kolodziejski, C. Dun, W. Huang, D. Carroll and S. M. Geyer, *Nano Res.*, 2017, **11**, 762–768.
- 26 H. Huang, B. Chen, Z. Wang, T. F. Hung, A. S. Sussha, H. Zhong and A. L. Rogach, *Chem. Sci.*, 2016, **7**, 5699–5703.
- 27 Z. J. Li, E. Hofman, J. Li, A. H. Davis, C. H. Tung, L. Z. Wu and W. Zheng, *Adv. Funct. Mater.*, 2018, **28**, 1704288.
- 28 Y. Cai, L. Wang, T. Zhou, P. Zheng, Y. Li and R. J. Xie, *Nanoscale*, 2018, **10**, 21441–21450.
- 29 V. Malgras, S. Tominaka, J. W. Ryan, J. Henzie, T. Takei, K. Ohara and Y. Yamauchi, *J. Am. Chem. Soc.*, 2016, **138**, 13874–13881.
- 30 H. C. Yoon, S. Lee, J. K. Song, H. Yang and Y. R. Do, *ACS Appl. Mater. Interfaces*, 2018, **10**, 11756–11767.
- 31 H. C. Wang, S. Y. Lin, A. C. Tang, B. P. Singh, H. C. Tong, C. Y. Chen, Y. C. Lee, T. L. Tsai and R. S. Liu, *Angew. Chem., Int. Ed. Engl.*, 2016, **55**, 7924–7929.
- 32 S. Guarnera, A. Abate, W. Zhang, J. M. Foster, G. Richardson, A. Petrozza and H. J. Snaith, *J. Phys. Chem. Lett.*, 2015, **6**, 432–437.
- 33 J. Y. Sun, F. T. Rabouw, X. F. Yang, X. Y. Huang, X. P. Jing, S. Ye and Q. Y. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1704371.
- 34 T. Altantzis, E. Coutino-Gonzalez, W. Baekelant, G. T. Martinez, A. M. Abakumov, G. V. Tendeloo, M. B. Roefsaers, S. Bals and J. Hofkens, *ACS Nano*, 2016, **10**, 7604–7611.
- 35 L. Protesescu, S. Yakunin, M. I. Bodnarchuk, F. Krieg, R. Caputo, C. H. Hendon, R. X. Yang, A. Walsh and M. V. Kovalenko, *Nano Lett.*, 2015, **15**, 3692–3696.
- 36 L. Wu, H. Hu, Y. Xu, S. Jiang, M. Chen, Q. Zhong, D. Yang, Q. Liu, Y. Zhao, B. Sun, Q. Zhang and Y. Yin, *Nano Lett.*, 2017, **17**, 5799–5804.
- 37 W. Lv, L. Li, M. Xu, J. Hong, X. Tang, L. Xu, Y. Wu, R. Zhu, R. Chen and W. Huang, *Adv. Mater.*, 2019, **31**, e1900682.
- 38 X. Zhang, X. Bai, H. Wu, X. Zhang, C. Sun, Y. Zhang, W. Zhang, W. Zheng, W. W. Yu and A. L. Rogach, *Angew. Chem., Int. Ed. Engl.*, 2018, **57**, 3337–3342.
- 39 Y. Wei, Z. Cheng and J. Lin, *Chem. Soc. Rev.*, 2019, **48**, 310–350.
- 40 X. Gong, M. Li, X. B. Shi, H. Ma, Z. K. Wang and L. S. Liao, *Adv. Funct. Mater.*, 2015, **25**, 6671–6678.

