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# Multimodal hybrid 2D networks *via* the thiol-epoxide reaction on 1T/2H MoS<sub>2</sub> polytypes†

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Searching for new protocols aimed at the surface functionalization of TMDCs is the password for filling the gap between reactivity and versatility of 2D carbon-based networks and their inorganic counterparts. This contribution describes a straightforward but effective approach to the chemical decoration of chemically exfoliated MoS<sub>2</sub> nanosheets with β-hydroxyl organic pendant arms. The adopted strategy is based on the thiol-epoxide “click” reaction and generates functional hybrids with a loading of dangling organic groups directly proportional to the epoxide electrophilic character. Most importantly, this grafted methodology occurs for the first time on both phases (1T and 2H) of the model chalcogenide and provides functional samples suitable for an additional functionalization step-forward (post-derivatization).

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## Introduction

The interest of the scientific community in organic or inorganic two-dimensional (2D) materials is growing exponentially year-by-year as their distinctive chemico-physical, optical and mechanical properties are often functional to address several technological issues.<sup>1–4</sup> Layered 2D transition-metal dichalcogenides (TMDCs) have already shown their potentiality in various technological areas as energy storage materials, components for nano-optoelectronic, sensing and environmental protection devices up to their application in catalysis.<sup>5–11</sup> Molybdenum disulfide (MoS<sub>2</sub>) is among the most studied material from the TMDCs series. It is available in several polymorphic structures classified on the basis of metal coordination, the most common ones being semiconducting 2H (trigonal prismatic) and metallic 1T (octahedral) phases.<sup>12,13</sup> The former represents the natural and thermodynamically stable MoS<sub>2</sub> polytype.<sup>14,15</sup> It can be chemically exfoliated by metal ion intercalation (using mainly Li<sup>+</sup> but also

Na<sup>+</sup> or K<sup>+</sup> ions), thus inducing a material structural reorientation towards the metallic 1T polymorph.<sup>16,17</sup>

Anyhow, a full exploitation of the unique features of these inorganic bidimensional networks implies a step forward in the direction of bridging the gap with their more celebrated and studied organic counterpart, graphene. In this regard, the surface tailoring of electronic and chemico-physical properties of MoS<sub>2</sub>, including material processability, can be conveniently achieved through chemical functionalization.<sup>18</sup> Recent findings from the literature have unveiled how the covalent surface functionalization of TMDCs boosts the (photo)electrocatalytic properties of these materials in key processes at the heart of renewable energy technology such as the hydrogen evolution reaction (HER).<sup>19–21</sup> Some of us have reported on the synthesis of an efficient hybrid TMDC sample with improved HER performance by anchoring newly designed porphyrins bearing different functional groups. We showed how mildly acidic hydroxyl groups in dangling organic fragments facilitate a local proton release, hence fostering hydrogen production.<sup>22</sup> However, any step forward in the direction of tailored hybrid functional materials requires the development of effective and versatile functionalization protocols to provide multimodality to the nanomaterial surface.

In recent years, few protocols for the chemical grafting of organic species to TMDCs have been developed.<sup>23–26</sup> The first example of MoS<sub>2</sub> surface decoration was reported by Dravid and co-workers using organic thiols.<sup>27</sup> These authors and related successive studies<sup>28–32</sup> postulated a “ligand conjugation” of exogenous thiol molecules at the S-vacancies of coordinatively unsaturated Mo atoms. Later, Mc Donald and co-workers<sup>33</sup> unveiled the real MoS<sub>2</sub>–thiols interaction nature and

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demonstrated that exogenous thiols were not covalently grafted to 2D nanosheets, but they rather underwent a formal oxidation step to give the corresponding disulfides that were in turn simply physisorbed at the material surface.

The first real covalent grafting of organic fragments to TMDC surface has been described by Chhowalla and collaborators in a seminal contribution that appeared in 2015.<sup>34</sup> Their protocol relied on the generation of negatively charged 1T-MoS<sub>2</sub> flakes (prepared by chemical exfoliation of a bulk 2H-MoS<sub>2</sub> sample with <sup>n</sup>BuLi) to be reacted with organoiodide species, covalently binding organic dangling arms to the sulfur atoms of the inorganic network. In the same year, Backes and co-workers developed a similar approach for the covalent grafting of chemically-exfoliated 1T-MoS<sub>2</sub> using reactive diazonium salts as electrophile species.<sup>16</sup> More recently, Perez *et al.* have described a different strategy for the mild decoration of 2H-MoS<sub>2</sub> obtained by liquid-phase exfoliation (LPE) in 2-propanol/water with an ultrasonic probe.<sup>35,36</sup> These authors exploited the soft nucleophilicity of sulfur atoms for covalent bond formation with maleimide derivatives *via* Michael addition. The reaction occurred at room temperature in the presence of a base while preserving the 2H semiconducting polytype of the starting material. Despite these seminal outcomes, much work remains to be done for covering the existing gap with the functionalization of organic 1D and 2D networks and hence for boosting the applications of TMDCs. Moreover, the development of new functionalization strategies suitable for the chemical modification of both 1T and 2H phases still remains a challenging and highly desirable objective.

This paper describes an effective and straightforward covalent functionalization strategy for the TMDC nanosheets decoration under mild reaction conditions. Taking the advantage of the principle of the effective thiol-epoxide “click” protocol,<sup>37</sup> we have reacted the negatively charged 1T-MoS<sub>2</sub> with a series of epoxide derivatives generating  $\beta$ -hydroxyl nanohybrids upon anti-Markovnikov reactant's ring opening. The successful MoS<sub>2</sub> nanosheet functionalization has been demonstrated by infrared (FT-IR) and X-ray photoelectron (XPS) spectroscopy, while elemental analysis (EA) crossed with TGA-MS provided a quantitative estimation of the grafted groups loading. The facile MoS<sub>2</sub> surface engineering with different organic moieties along with the control on the dangling groups loading is a highly powerful tool for the fine-tuning of MoS<sub>2</sub> chemico-physical properties. It is noteworthy that the proposed covalent grafting approach takes place on both metallic (1T) and semiconducting (2H) MoS<sub>2</sub> phases. Finally, the hydroxyl moieties derived from the epoxides ring-opening have undergone further post-functionalization, thus conferring multimodality to MoS<sub>2</sub> platforms and widening their applicability range.

## Results and discussion

### Chemical functionalization of CE-MoS<sub>2</sub> by anti-Markovnikov epoxides ring-opening

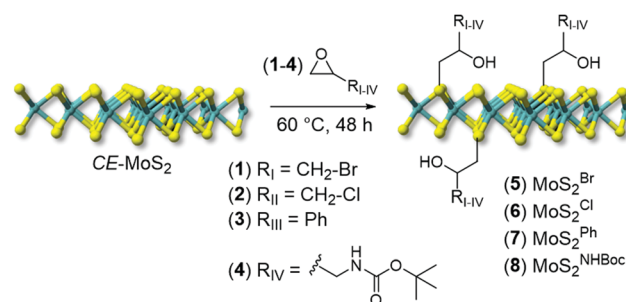
Few-layer 1T-MoS<sub>2</sub> nanosheets are prepared by Li-ion intercalation according to a literature procedure recently reported by some

of us.<sup>38</sup> In brief, a solid mixture of bulk 2H-MoS<sub>2</sub> and LiBH<sub>4</sub> was treated at 330 °C for 4 days and then transferred to degassed Milli-Q water. After sonication and careful purification, a stable and homogeneous aqueous dispersion of chemically-exfoliated MoS<sub>2</sub> (CE-MoS<sub>2</sub> ~ 0.2 mg mL<sup>-1</sup>) was obtained. According to the literature, XPS analysis of the exfoliated material (Fig. S1, ESI<sup>†</sup>) revealed a predominant 1T phase<sup>38</sup> (72% *vs.* 16% of 2H-MoS<sub>2</sub>) with only minor amounts of partially oxidized MoS<sub>x</sub>O<sub>y</sub> species (7%) and MoO<sub>x</sub> suboxides (5%).

Our approach to chemical decoration of CE-MoS<sub>2</sub> relies on the ability of the as-obtained negatively charged chalcogenide to react as a nucleophile towards epoxide derivatives, leading to their regioselective ring-opening *via* the so-called thiol-epoxide “click” reaction.<sup>37,39</sup> The latter is a one-step, selective and 100% atom economy process that takes place with high yield under mild operative conditions.<sup>40</sup> All these aspects make the protocol highly attractive for its exploitation as a new and efficient covalent functionalization method for the TMDC surface decoration with different chemical functionalities. Moreover, the hydroxyl moieties generated upon epoxide ring-opening are reactive sites that may undergo further post-functionalization, thus improving the versatility and widening the applicability range of this emerging class of 2D materials (*vide infra*).

A series of differently substituted epoxide derivatives (**1–4**) have been selected as model reactants and underwent nucleophilic attack from MoS<sub>2</sub> nanosheets (Scheme 1). In a typical procedure, 200 mL of CE-MoS<sub>2</sub> aqueous suspension reacted with a proper amount of epoxide (10 eq. respect to S) at 60 °C (see Experimental section for details).

After the materials work-up, first and qualitative evidence of the occurred epoxide ring-opening is provided by FT-IR spectra of all functionalized samples (**5–8**, Fig. 1). In particular, distinctive signals ascribed to alkane C–H stretching ( $\nu_{C-H}$ ) and an intense broad band unambiguously assigned to the stretching of hydroxyl groups ( $\nu_{O-H}$ ) appear for all decorated materials at ~2900 and ~3400 cm<sup>-1</sup>, respectively.<sup>41</sup> Noteworthy, **5** shows a remarkably less intense IR signal associated with –OH groups with respect to all other grafted materials (**5 vs. 6–8**, Fig. 1). Such evidence prompted us to invoke the occurrence of an alternative nucleophilic attack on epibromohydrin (**1**). Indeed, the presence of bromine as a good leaving group opens two different scenarios competitive with the plain anti-Markovnikov epoxide ring-



**Scheme 1** Chemical functionalization of CE-MoS<sub>2</sub> with epoxide derivatives (**1–4**) to give decorated MoS<sub>2</sub> nanohybrids (**5–8**).



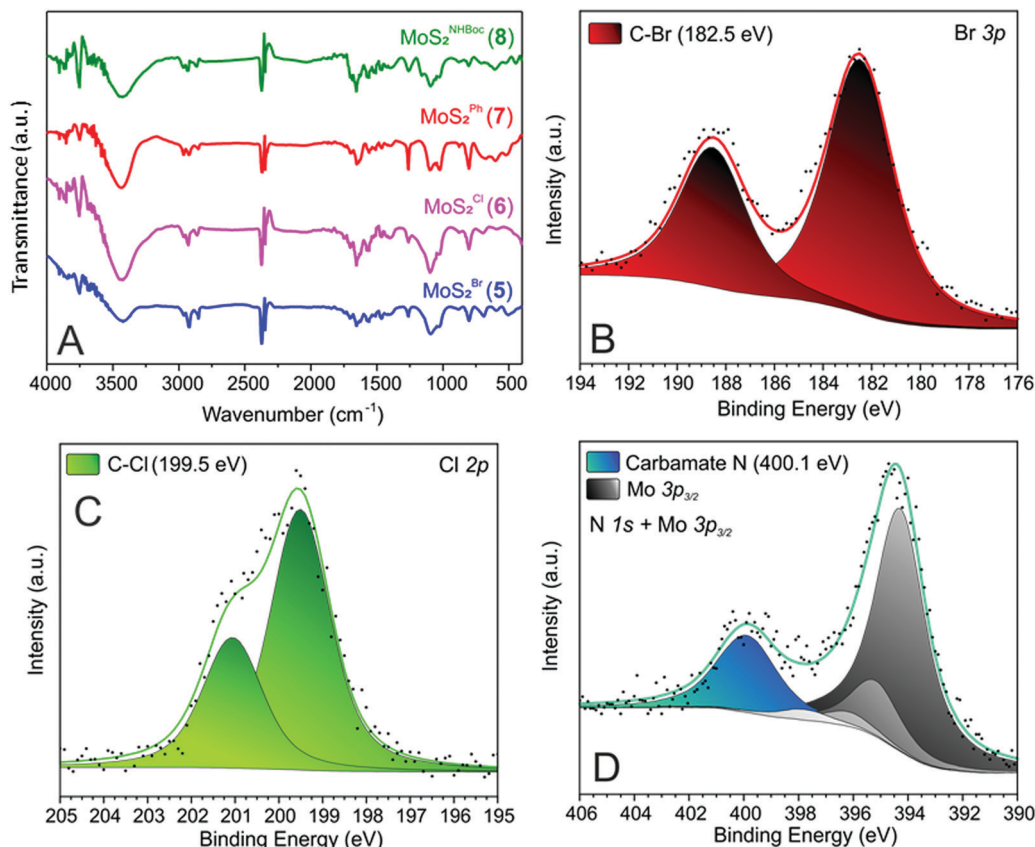


Fig. 1 (A) IR spectra of MoS<sub>2</sub> functionalized materials (5–8). (B) High-resolution Br 3p XPS core level region of 5. (C) High-resolution Cl 2p XPS core level region of 6. (D) High-resolution N 1s XPS core level region of 8.

opening: the direct S-nucleophilic displacement at the C–Br bond or a Payne rearrangement just after the initial epoxide ring-opening with the subsequent intramolecular ring-closure and oxirane regeneration.<sup>42–44</sup> In both cases, the final product would give rise to a CE-MoS<sub>2</sub> decorated sample bearing integral epoxide moieties instead of free-hydroxyl groups. Although these authors are aware of the qualitative character of the IR information, particularly for poorly distinctive signals like those arising from this region of the IR spectrum, the lower intensity of the IR band associated with –OH moieties for this sample (5) can be coherently rationalized as discussed above. Since the pioneering work by Chowalla and co-workers,<sup>34</sup> alkyl iodide derivatives are known to undergo nucleophilic substitution (S<sub>N</sub>) in the TMDC surface functionalization. On the other hand, such reactivity was considered less feasible in the presence of alkyl bromides. As an example, McDonald and co-workers reported on the covalent grafting of 5-bromomethyl-2,2'-bipyridine to CE-1T-MoS<sub>2</sub> in the presence of a catalytic amount of NaI in the reaction medium; the addition of NaI was necessary to allow the halogen scrambling on the organic reactant, with concomitant occurrence of the nucleophilic substitution reaction.<sup>45</sup> However, more recently it has been demonstrated that alkyl bromide derivatives can also be exploited for the direct covalent decoration of CE-MoS<sub>2</sub> via the S<sub>N</sub> reaction,<sup>15,46</sup> supporting our hypothesis of competing processes at work with epibromohydrin (1) as a substrate.

The signal around 3400 cm<sup>−1</sup> in 5 (even if less intense with respect to the other hybrid materials 6–8, Fig. 1) suggests indeed that the epoxide ring-opening (partially) occurs also with 1. The presence of bromine in 5 (as evidenced by TGA-MS and XPS analyses – *vide infra*) provides additional evidence for the existence of two alternative reaction paths at work with 1 as a substrate.

Further proof of evidence of the occurred CE-MoS<sub>2</sub> functionalization comes from XPS analyses. All MoS<sub>2</sub> nanohybrids 5–8 show distinctive signals associated with heteroatoms at the respective grafted moieties. Binding energy (BE) signals at 182.5 and 199.5 eV for Br 3p and Cl 2p core regions of 5 and 6, respectively (Fig. 1B and C) account for the occurred nanomaterial grafting and epoxide ring-opening on oxiranes 1 and 2. Similarly, a distinctive N 1s signal ascribed to the carbamate moiety<sup>47</sup> (400.1 eV, Fig. 1D) is observed in the XPS spectrum of 8.

As far as distinctive XPS signals of 7 are concerned, the high-resolution C 1s spectrum (Fig. S2C, ESI†) is in accordance with the styrene oxide (3) ring-opening as outlined in Scheme 1. Indeed, two components at 284.5 and 286.9 eV with a roughly 3 : 1 intensity ratio and ascribed to phenyl and C–OH/C–S moieties, respectively, are clearly observed in the C 1s photoemission spectrum of 7.

Finally, the shift towards higher BEs of the S 2p XPS signal in all grafted 2D nanomaterials (indicating an electronic transfer from the inorganic MoS<sub>2</sub> nanosheets to its covalently grafted organic arms) is an indirect evidence of the occurred surface



MoS<sub>2</sub> nanosheets decoration. Noteworthy, both 1T and 2H phases underwent a significant shift (1T: from 160.9 eV in the pristine CE-MoS<sub>2</sub> to 161.1–161.5 eV in 5–8; 2H: from 161.9 eV in the pristine CE-MoS<sub>2</sub> to 162.2–162.5 eV in 5–8, Fig. S1 vs. S2–S5, ESI†). Such evidence suggests that both metallic and semiconducting MoS<sub>2</sub> polymorphs equally underwent covalent grafting. This aspect is of primary importance because the 2H phase cannot be directly functionalized by strong electrophiles<sup>48</sup> such as alkyl halides<sup>34</sup> or aryldiazonium salts<sup>16,38</sup> (if not in the presence of pre-formed defects as nucleation sites).<sup>49</sup> To the best of our knowledge, the only example of a direct 2H MoS<sub>2</sub> functionalization reported so far, deals with the recent contribution by Perez and co-workers and it is limited to TMDCs decoration with maleimide derivatives only.<sup>35</sup> Thereby, the thiol-epoxide functionalization offers a valid and practical alternative to existing methods for surface grafting of both metallic and semiconducting MoS<sub>2</sub> phases using a wide variety of functional groups. A careful analysis of the S 2p photoemission spectra of 5–8 (Fig. S1 vs. S2–S5, ESI†) has finally unveiled that 5 and 6 present more pronounced signal shifts (~0.5–0.6 eV vs. plain MoS<sub>2</sub>) towards higher BEs compared to those measured on the two other nanohybrids (7–8; ~0.2–0.4 eV vs. plain MoS<sub>2</sub>). This trend mirrors the electronic properties of the material decorating groups: the higher the electron-withdrawing character of the pendant arms, the higher the electronic transfer from MoS<sub>2</sub> nanosheets to its grafted species and thus the higher the BE values of the corresponding S 2p XPS signals. On the other hand, Mo 3d XPS photoemission spectra of all functionalized derivatives (5–8) did not change significantly with respect to the starting CE-MoS<sub>2</sub> (Fig. S1 vs. S2–S5, ESI†). Such an effect indicates that electronic material properties are not altered

(also in terms of 1T/2H phases ratio) by the thiol-epoxide “click” process. Overall, XPS analyses demonstrate that chemical functionalization occurs at negatively charged sulfur atoms, in line with their ability to react as nucleophiles towards epoxide derivatives.

On a more quantitative ground, thermogravimetric (TGA-MS, Fig. 2) and elemental analyses (Table 1) have been accomplished on samples 5–8 to better determine the functionalization loading of each nanohybrid. Weight losses measured on 5–8 have been compared with the TGA profile of pristine CE-MoS<sub>2</sub> to provide direct and semi-quantitative evidence of the organic groups loading grafted on the inorganic 2D network. At the same time, the MS analysis of volatiles produced throughout the material's thermal decomposition has been used to confirm the identity of the chemically grafted functionalities.

As Fig. 2 shows, samples 5 and 7 display distinctive weight losses in the 60–300 °C range, whose net decrease reduced by that of the plain CE-MoS<sub>2</sub> in the same temperature range (10.9 wt% for 5 and 5.9 wt% for 7) is roughly attributed to the complete thermal decomposition of organic moieties grafted to the material surface. MS analyses of low-molecular-weight volatiles produced throughout nanohybrids thermal decomposition additionally support the nature of the MoS<sub>2</sub> grafted species.

While distinctive mass peaks attributed to HBr ( $m/z = 80$  and  $82$  [ $M^+$ ] coming from <sup>79</sup>Br and <sup>81</sup>Br isotopes, respectively) together with less intense components due to CH<sub>2</sub>-Br species ( $m/z = 93$  and  $95$  [ $M^+$ ]) were observed for 5 (Fig. 2A), mass signals ascribable to phenyl C<sub>6</sub>H<sub>5</sub> ( $m/z = 77$  [ $M^+$ ],  $m/z = 78$  [ $M^+ + 1$ ]) and styrene Ph-CH=CH<sub>2</sub> moieties ( $m/z = 104$  [ $M^+$ ],  $m/z = 105$  [ $M^+ + 1$ ] and  $m/z = 103$  [ $M^+ - 1$ ]) were detected for 7 (Fig. 2C). Similarly,

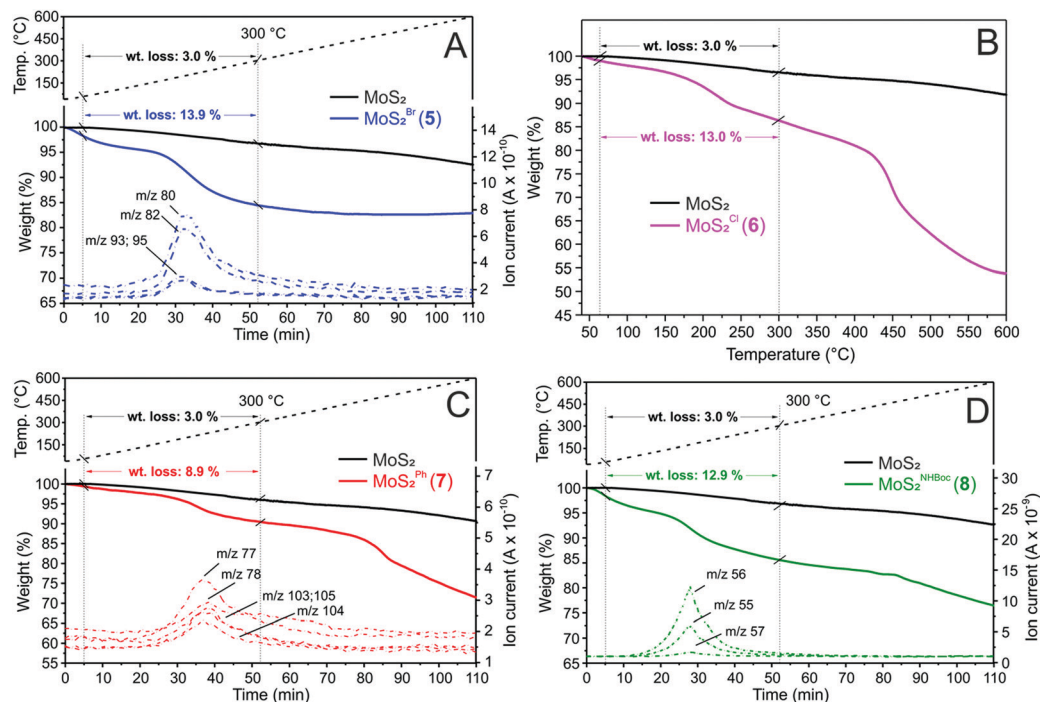


Fig. 2 Thermogravimetric profiles of (A) MoS<sub>2</sub><sup>Br</sup> (5), (B) MoS<sub>2</sub><sup>Cl</sup> (6), (C) MoS<sub>2</sub><sup>Ph</sup> (7) and (D) MoS<sub>2</sub><sup>NHoc</sup> (8) along with (whenever applicable) MS analysis of volatiles. Thermal program: 40–600 °C, 5 °C min<sup>-1</sup>, N<sub>2</sub> atmosphere (50 mL min<sup>-1</sup>).





**Table 1** Elemental analysis (EA) and estimated functional groups loading for samples **5–8**

Samples	Elemental analysis <sup>a</sup>			S/R <sub>I-IV</sub> molar ratio	R <sub>I-IV</sub> -loading (mmol g <sup>-1</sup> )	
	N wt%	C wt%	S wt%		From EA <sup>b</sup>	From TGA <sup>c</sup>
CE-MoS <sub>2</sub>	—	1.63	23.19	—	—	— <sup>d</sup>
MoS <sub>2</sub> <sup>Br</sup> ( <b>5</b> )	—	8.08	22.61	3.9	1.79	—
MoS <sub>2</sub> <sup>Cl</sup> ( <b>6</b> )	—	5.91	26.23	6.9	1.19	1.07
MoS <sub>2</sub> <sup>Ph</sup> ( <b>7</b> )	—	8.99	23.41	9.5	0.77	0.49
MoS <sub>2</sub> <sup>NHBoc</sup> ( <b>8</b> )	1.31	9.57	27.39	10.3	0.83 (0.93) <sup>e</sup>	0.57

<sup>a</sup> Calculated as average values over three independent runs. <sup>b</sup> Loading calculated from C wt%. <sup>c</sup> Calculated from TG weight losses (%) in the 60–300 °C temperature range. <sup>d</sup> Not determined because of secondary reaction paths at work (*vide infra*). <sup>e</sup> Loading calculated from N wt%.

a characteristic weight loss (12.9 wt%) was registered for sample **8** in the same 60–300 °C temperature range (~5–52 min). The mass analysis of volatiles highlights the generation of isobutene ( $m/z = 56$  [ $M^+$ ],  $m/z = 55$  [ $M^+ - 1$ ] and  $m/z = 57$  [ $M^+ + 1$ ]), typically ascribed to the thermal decomposition and rearrangement of *tert*-butoxyl groups of the N-Boc protected carbamate species (Fig. 2D).<sup>38,47</sup> Finally, **6** shows a clear-cut weight loss of 13.0 wt% in the 60–300 °C temperature range (Fig. 2B) that accounts for 1.1 mmol g<sup>-1</sup> of  $\beta$ -hydroxyl chloride-grafted moieties. In this case, the low molecular weight of volatiles evolved throughout the material thermal decomposition did not allow the detection of any characteristic mass signals.

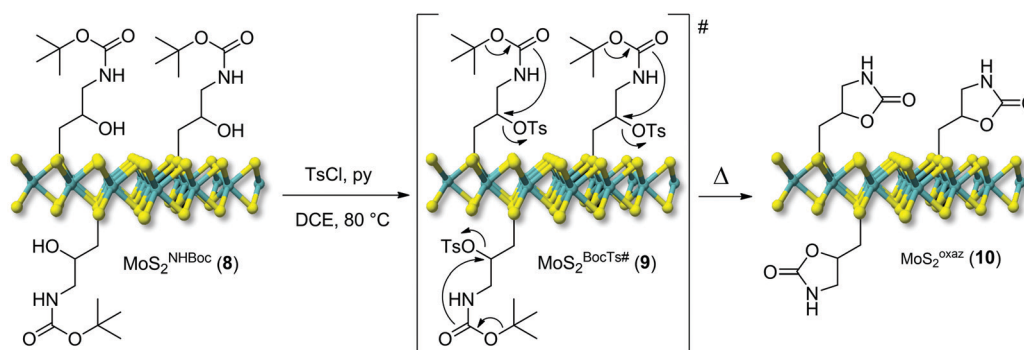
Noteworthy, we found an excellent match between the functionalization degree extrapolated from TGA analyses and the organic fraction loading calculated from EA (Table 1).

The loading of grafted groups for nanohybrids **7** and **8** is quite similar with roughly one (**1**) decorating moiety every 9.5 and 10.3 sulfur atoms, respectively (as calculated from EA analyses, Table 1). Such a percentage slightly increases on **6** where a S/R<sub>II</sub> ratio close to 6.9 is calculated. This trend is in line with the higher electrophilic character of epichlorohydrin (**2**) that fosters the thiol-epoxide “click” reaction with respect to epoxides **3** and **4**, thus enhancing the MoS<sub>2</sub> surface coverage.<sup>19</sup> On the other hand, the markedly higher functionalization degree measured on **5** (S/R<sub>II</sub> = 3.9) cannot simply be ascribed to the electron-withdrawing nature of bromine but rather to the occurrence of a competitive bromide displacement *via* direct SN (or Payne rearrangement – see above). In addition, the weight

loss measured by TGA analyses on **5** (13.9%, Fig. 2A) is markedly lower with respect to what we expected for a pure thiol-epoxide ring-opening functionalization (~28%). Such a discrepancy additionally supports the hypothesis for a competing SN process at work in the case of epoxide **1**. Crossing EA outcomes with thermogravimetric weight loss measured for the nanohybrid **5**, it can be inferred that such a secondary reaction path largely prevails on the thiol-epoxide “click” reaction (estimated SN: thiol-epoxide “click” ratio ~75:25). Worthy of note, this ratio well-matches with that of 1T/2H phases of the starting CE-MoS<sub>2</sub> (72/16, see Fig. S1, ESI†). The higher chemical inertness of the semiconducting 2H phase towards direct decoration in the presence of strong electrophiles (*i.e.* alkyl halides)<sup>16,34,38</sup> has prompted us to postulate that the minor 2H MoS<sub>2</sub> component is the only phase involved in the thiol-epoxide “click” reaction with epibromohydrin (**1**), while the major metallic 1T-MoS<sub>2</sub> phase undergoes classical SN reaction with bromide ion elimination.

### Post-synthetic modification of nanohybrids **7** and **8**

One key challenge in the chemical functionalization of 2D organic/inorganic platforms is represented by the possibility to impart multimodality to these layered materials. The thiol-epoxide protocol is particularly relevant because the oxirane ring-opening provides at least one highly versatile hydroxyl group suitable to undergo tailored conversion/derivatization. On this ground, the nanohybrid **8** was treated with *p*-toluenesulfonyl chloride (CH<sub>3</sub>C<sub>6</sub>H<sub>4</sub>SO<sub>2</sub>Cl) and pyridine (C<sub>5</sub>H<sub>5</sub>N) as an acid scavenger in refluxing dichloroethane (DCE, C<sub>2</sub>H<sub>4</sub>Cl<sub>2</sub>) for the hydroxyl moiety conversion into an optimal leaving-group. All attempts to carry out the reaction at either room temperature or refluxing temperature of dichloromethane (CH<sub>2</sub>Cl<sub>2</sub>, bp 40 °C) failed and the nanohybrid **8** was entirely recovered with no alterations. As expected for the intermediate **9**, the transiently formed tosyl-ester undergoes spontaneous intramolecular cyclization, thus leading to the oxazolidinones-decorated nanohybrid **10** (MoS<sub>2</sub><sup>oxaz</sup>, Scheme 2).<sup>50,51</sup> Such reactivity of the organyl pendant fragment is favored already at 0 °C in the case of N-alkyl substituted carbamates, while it occurs upon heating at about 60 °C on NH derivatives. After a careful material workup, sample **10** was analyzed first using TGA-MS. As Fig. 3A shows, MoS<sub>2</sub><sup>oxaz</sup> (**10**) shows a lower weight loss (10.1 wt%) if compared with the starting material (**8**) in the same temperature range. Moreover, the weight loss measured

**Scheme 2** Post-derivatization of the nanohybrid MoS<sub>2</sub><sup>NHBoc</sup> (**8**) into the oxazolidinones-decorated MoS<sub>2</sub><sup>oxaz</sup> (**10**).

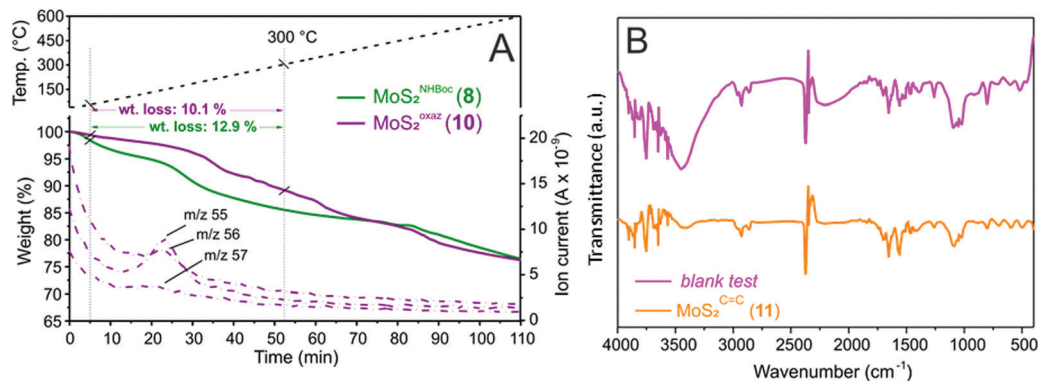


Fig. 3 (A) Thermogravimetric profiles of  $\text{MoS}_2^{\text{oxaz}}$  (**10**) along with mass analysis of volatile species. Thermal program: 40–600 °C, 5 °C min<sup>-1</sup>, N<sub>2</sub> atmosphere (50 mL min<sup>-1</sup>). TG profile of  $\text{MoS}_2^{\text{NHBOC}}$  (**8**) is reported for the sake of comparison. (B) IR spectrum of  $\text{MoS}_2^{\text{C}=\text{C}}$  (**11**) along with the relative blank test.

for the nanohybrid **10** is in excellent accord with that expected for a complete conversion of tosylated  $\beta$ -hydroxyl moieties in **8** ( $\sim 0.7 \text{ mmol g}^{-1}$ , Table 1) into the corresponding oxazolidinone species (10.0 wt%).

Further proof of evidence of the occurred intramolecular cyclization is given from MS analysis of low-molecular-weight volatiles resulting from the thermal decomposition of **10** (Fig. 3A). MS signals ascribable to the oxazolidinone thermal rearrangement into 2-methyl-aziridine species ( $m/z = 56 [\text{M}^+]$ ,  $m/z = 55 [\text{M}^+ - 1]$ )<sup>52</sup> along with complete suppression of the mass peaks associated with the thermal rearrangement of *tert*-butoxyl groups in the Boc units of the starting material **8** (Fig. 2D), are useful additional tiles of a complex puzzle that validate the post-synthetic treatment of **8** as well as the supposed nature of the decorating fragments in **10**.

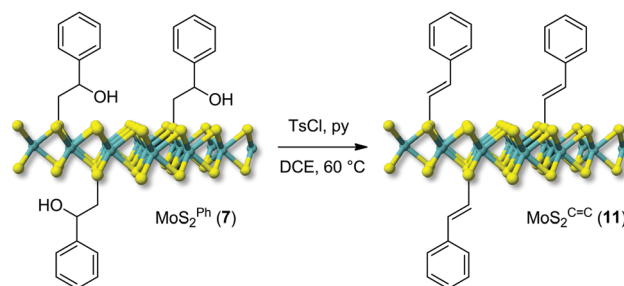
Elemental analysis on  $\text{MoS}_2^{\text{oxaz}}$  (**10**) has finally confirmed the occurred post-derivatization path (Table S1, ESI†). Indeed, a coherent reduction of C wt% was recorded from the nanohybrid **8** to its post-derivative **10**, while the estimated functionalization degree remained almost unchanged ( $\sim 0.83 \text{ mmol g}^{-1}$ ; 1 group every 9.7 sulfur atoms).

IR spectra of hybrids **8** and **10** at comparison show a marked decrease of the  $-\text{OH}$  groups stretching ( $\nu_{\text{O-H}}$ ;  $\approx 3400 \text{ cm}^{-1}$ ) on the latter (Fig. S6, ESI† vs. Fig. 1), while the N 1s photoemission spectrum of **10** presents a unique component centered at 400.0 eV (Fig. S7, ESI†) ascribed to a carbamate fragment. Both results are in line with the expected generation of oxazolidinone species on the 2D nanomaterial surface. At the same time, no appreciable variation of Mo 3d or S 2p photoemission lines in two nanohybrids at comparison (**8** vs. **10**) is observed (Fig. S5 vs. S7, ESI†), which confirms that the post-synthetic modification on **8** occurs without any appreciable alteration of the  $\text{MoS}_2$  pristine structure.

Finally, the reactivity of hydroxyl groups in the nanohybrid **7** was tested under similar post-derivatization conditions, using *p*-toluenesulfonic acid and pyridine as an acid scavenger. In this case, the treatment resulted in the formal dehydration of the tosylated secondary hydroxyl group with the subsequent generation of a conjugated  $\text{C}=\text{C}$  bond ( $\text{MoS}_2^{\text{C}=\text{C}}$  **11**, Scheme 3).<sup>53,54</sup>

The first evidence of the occurred alcohol dehydration is given by IR analysis of derivative **11**. As shown in Fig. 3B, the  $-\text{OH}$  stretching ( $\nu_{\text{O-H}} \sim 3400 \text{ cm}^{-1}$ ) is markedly reduced on **11** compared to the starting precursor **7**. To get further evidence of the occurred secondary alcohol dehydration, a blank trial has been carried out using **7** as a substrate to be reacted under identical conditions to those outlined on Scheme 3 except for the use of the *p*-toluenesulfonic acid. Noteworthy, the IR spectrum of the blank sample was largely superimposable on that of the starting hybrid **7** with no appreciable suppression of the stretching signal attributed to the secondary alcohol (Fig. 3B). All these data taken together strongly supported the hypothesis of an alcohol elimination path facilitated by the transient generation of the sulfonyl ester intermediate.

Thermogravimetric analysis conducted on  $\text{MoS}_2^{\text{C}=\text{C}}$  (**11**) was also consistent with the postulated alcohol dehydration path. Indeed, the nanohybrid **11** showed only a slightly lower weight loss with respect to the starting material **7** (8.4 wt% vs. 8.9 wt%) while keeping almost unchanged its TG profile all over the whole scanned temperature range (Fig. S8, ESI†). Similarly, elemental analysis of **11** (Table S1, ESI†) did not show any appreciable alteration with respect to **7**, thus strengthening the supposed dehydration path. As a matter of fact, the functionalization degree calculated for hybrid **11** was in excellent accord with the loading measured on its starting material **7** (Table S1, ESI† vs. Table 1).



Scheme 3 Post-derivatization of the nanohybrid  $\text{MoS}_2^{\text{Ph}}$  (**7**) into the styrene-decorated  $\text{MoS}_2^{\text{C}=\text{C}}$  (**11**).



Finally, XPS analysis was used to corroborate the expected formal alcohol dehydration mechanism at work on **7**. In particular, the C 1s core-level regions of the two nanohybrids in comparison (**7** vs. **11**; Fig. S2 vs. S9, ESI†) have unveiled a coherent increment in **11** of the component at 284.4 eV ascribed to C sp<sup>2</sup> moieties along with a corresponding decrease of the peak at 286.9 eV attributed to C–OH groups.

## Experimental

### General considerations

All manipulations were carried out under dry nitrogen ( $\geq 99.999\%$ ; Rivoira) atmosphere using standard Schlenk-type techniques. Aqueous suspension of CE-MoS<sub>2</sub> ( $\sim 0.2$  mg mL<sup>-1</sup>) was obtained by following literature procedures.<sup>38,55,56</sup> Unless otherwise stated, all chemicals were provided by commercial suppliers and used as received. Sample sonications were conducted on an Elma S15 Elmasonic sonicator (37 kHz) in a water/ice bath. Thermogravimetric analysis (TGA-MS) was run under N<sub>2</sub> atmosphere (50 mL min<sup>-1</sup>) on an EXSTAR Seiko 6200 analyzer coupled with a ThermoStar™ GSD 301T for mass analysis of volatile species. All samples (from 10 to 40 mg) were stabilized to constant weight at 40 °C (from 30 min to 90 min) before running each measurement. Elemental analysis was carried out on a Thermo FlashEA 1112 series CHNS-O elemental analyzer with an accepted tolerance of  $\pm 2\%$  on each element. N, C and S wt% were determined as mean values over three independent analyses. FT-IR spectroscopy was performed on a PerkinElmer Spectrum BX FT-IR spectrophotometer in the 400–4000 cm<sup>-1</sup> range with a resolution of 1 cm<sup>-1</sup>. Samples for analyses were prepared by mixing spectroscopic-grade KBr with functionalized MoS<sub>2</sub> hybrids. X-Ray photoemission spectroscopy (XPS) measurements were performed under UHV conditions ( $< 10^{-8}$  mbar) at room temperature using a nonmonochromated Al K $\alpha$  source ( $h\nu = 1486.7$  eV) and an electron analyzer pass energy of 20 eV. The binding energy (BE) scale was calibrated using Au 4f<sub>7/2</sub> core levels (BE = 83.8 eV). XPS multipeak analysis (KolXPd software)<sup>57</sup> was performed using Voigt functions, keeping constant in all the peaks the full width at half maximum (FWHM) and the Gaussian–Lorentzian ratio.

### General procedure for the synthesis of MoS<sub>2</sub> functionalized samples 5–8

200 mL of an aqueous suspension of CE-MoS<sub>2</sub> ( $\sim 0.2$  mg mL<sup>-1</sup>) is treated with a proper amount of epoxide derivative (5.5 mmol, 10 eq. vs. S) and sonicated for 20 minutes. Afterward, the suspension was stirred at 60 °C for 48 h, cooled to room temperature and centrifuged in order to recover the solid residue. The latter was washed/sonicated several times (3  $\times$  30 mL of EtOH and 3  $\times$  30 mL of AcOEt, 15 min sonication each) and recovered by centrifugation before being dried to constant weight and stored at room temperature.

### Post-derivatization of MoS<sub>2</sub><sup>Ph</sup> (**7**) and MoS<sub>2</sub><sup>NHBoc</sup> (**8**)

20 mg of functionalized MoS<sub>2</sub> [either MoS<sub>2</sub><sup>Ph</sup> (**7**) or MoS<sub>2</sub><sup>NHBoc</sup> (**8**)] were suspended in 20 mL of dry and degassed dichloroethane

(DCE) and sonicated for 20 minutes. Afterwards, an excess of tosyl chloride (TsCl, 5 eq. respect to hydroxyl group loading calculated on the basis of EA – 0.07 and 0.08 mmol for hybrids **7** and **8**, respectively) and pyridine (25 eq.) was added to the suspension. After sonication for further 10 minutes, the mixture was maintained under stirring at 50 °C (80 °C for hybrid **8**) for 3 days. After cooling to room temperature, the solid was recovered by centrifugation and subjected to several purification cycles (sonication/centrifugation) with ethyl acetate (3  $\times$  30 mL) and EtOH (3  $\times$  30 mL) before being dried to constant weight. For the blank test, sample **7** was treated exactly under the same reaction conditions except for the use of *p*-toluenesulfonic acid as a reagent.

## Conclusions

We described an efficient and general strategy for the covalent grafting of organic moieties to chemically exfoliated MoS<sub>2</sub> nanosheets, taking advantage of the thiol-epoxide “click” protocol. To accomplish this aim, exfoliated MoS<sub>2</sub> nanoflakes have been reacted under relatively mild conditions with a series of epoxide derivatives from a wide collection of commercially available compounds, giving rise to chemically decorated 2D MoS<sub>2</sub> networks containing  $\beta$ -hydroxyl pendant arms. Besides, we have demonstrated the efficiency of the proposed functionalization technology for the material surface decoration that occurs on both semiconducting and metallic phases of CE-MoS<sub>2</sub>. To the best of our knowledge, this is the first example reported so far wherein covalent grafting of organic moieties at the inorganic surface of a model chalcogenide occurs at both 1T and 2H phases. Furthermore, the grafting approach generates reactive hydroxyl surface groups that may undergo further derivatization through classical post-functionalization procedures. Such a feature fills the existing gap between the reactivity of classical C-based 1D and 2D networks and that of 2D TMDCs.

All materials were thoroughly characterized by different and complementary spectroscopic (IR, XPS) and analytical (Elemental analysis and TGA-MS) techniques, along with (whenever applicable) blank trials. The quantitative estimation of the grafted groups has unveiled a correlation between the material functionalization degree and electrophilic character of the epoxide derivatives dictated by the electron-withdrawing character of their substituents. Overall, these findings widen the toolkit of TMDC functionalization protocols, providing multimodality to the surface of the resulting materials. This aspect opens new horizons in the exploitation of hybrid TMDCs as (photo)electrocatalysts in processes where they have already shown excellent performances. Hybrid hetero-tagged TMDC derivatives of this type are currently being investigated in our labs as functional carriers for (photo)electrocatalytic active moieties in HER.

## Conflicts of interest

There are no conflicts to declare.





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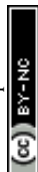
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