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Preparation of well-defined 2D-lenticular aggregates by self-assembly of PNIPAM-*b*-PVDF amphiphilic diblock copolymers in solution†

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PNIPAM-*b*-PVDF (poly(*N*-isopropylacrylamide)-*b*-poly(vinylidene fluoride)) amphiphilic block copolymers (BCPs) were synthesized *via* RAFT polymerization from PNIPAM macromolecular chain transfer agents (macro-CTAs). PNIPAM was used as it is one of the few hydrophilic polymers soluble in dimethyl carbonate (DMC) and which can be synthesized using a xanthate CTA with relatively good control. The polymerizations were conducted at 73 °C in DMC using two PNIPAM macro-CTAs of different molar masses and targeting various DPs for the PVDF block. The RAFT polymerization of VDF resulted in relatively well-defined BCPs ($Đ \leq 1.50$). These amphiphilic BCPs were able to self-assemble into various morphologies such as spherical, crumpled, lamellar and lenticular 2D aggregates by changing the common solvent or the self-assembly protocol. The size of the aggregates could be controlled by varying the DP of the PVDF block. The polymers were characterized by ^{1}H and ^{19}F NMR, SEC, TGA and DSC, and the assembled structures were studied by TEM, SEM and AFM. The thermosensitive behavior of the 2D lenticular aggregates was also examined.

Introduction

Poly(vinylidene fluoride) (PVDF), despite its remarkable properties such as piezoelectricity, ferroelectricity, chemical inertness, and biocompatibility,[1,2] has not received as much attention as other polymers in fields such as macromolecular engineering and polymer self-assembly. This is likely due to the synthesis constraints inherent to the peculiar reactivity of VDF. In addition, since VDF is a gaseous monomer, its polymerization usually requires the use of a high-pressure vessel, not commonly available in every laboratory. However, in recent years, the use of reversible-deactivation radical polymerization techniques to prepare PVDF and PVDF-based BCPs has been investigated with some success.[3–6]

RAFT polymerization and iodine-transfer polymerization of VDF lead to the accumulation of VDF tail-terminated chains

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($-CF_2CF_2CH_2-X$, X = xanthate or iodine).[7] These chains are not easily reactivated which hinders the preparation of PVDF-based block copolymers. So far, only PVDF-*b*-PVAc block copolymers (VAc = vinyl acetate) could be synthesized by chain extension of CF_2-CH_2-XA-terminated–PVDF (XA = xanthate) prepared by RAFT polymerization.[8] Cobalt-mediated radical polymerization was shown to largely mitigate this end-group reactivity issue, and is arguably the best RDRP technique for VDF.[4,9] However, it requires the use of air-sensitive organometallic compounds and is thus not easy to implement. Although coupling strategies using click chemistry (CuAAC, or thia-Michael addition for example)[10–13] are efficient, they may require the preparation of functional RAFT agents, and often lead to mixtures of block copolymers and homopolymers which are not easy to separate. Perhaps the most straightforward solution is the chain extension of RAFT macromolecular chain transfer agents (macro-CTAs) with PVDF.

To date, only four reports have described the preparation of PVDF-based block copolymers from RAFT macro-CTAs using sequential addition of VDF. Kostov *et al.* described the synthesis of PVAc-*b*-P(VDF-*co*-TFP) block copolymers (TFP = trifluoropropene).[14] Girard *et al.* reported the preparation of PDMA-*b*-PVDF and PDMA-*b*-P(VDF-*co*-PMVE) by chain extension of a PDMA macro-CTA (DMA = dimethylacrylamide, PMVE = perfluorinated methyl vinyl ether).[15] Guerre *et al.* reported the preparation of PEVE-*b*-PVDF (EVE = ethyl vinyl

ether) block copolymers *via* the sequential combination of cationic RAFT polymerization of vinyl ethers and radical RAFT polymerization of VDF.[16] Guerre *et al.* also reported the polymerization-induced self-assembly (PISA) of PVAc-*b*-PVDF block copolymers.[17] VAc units were then hydrolyzed to prepare PVA-*b*-PVDF amphiphilic BCPs (VA = vinyl alcohol).[18] Only PDMA- and PVA-based PVDF-containing amphiphilic block copolymers have been prepared by sequential addition of VDF.

PVDF is a semi-crystalline fluorinated polymer, which as such is appealing for the preparation of self-assembled block copolymer morphologies through crystallization-driven self-assembly (CDSA) in selective solvents. The self-assembly of non-crystalline (coil–coil) BCPs in solution is well established.[19] However, the complexity of the self-assembly process increases when one block of the BCP can crystallize. The formation of semi-crystalline BCP self-assembled structures can be viewed as a two-step process. The structures will first form by minimizing the unfavourable contacts between the solvophobic block and the solvent, and the semi-crystalline block will then start to crystallize in a second step, leading to the final structure. As crystallization takes place in the insoluble core, the initial morphology is either preserved or a morphological transformation into a novel structure is triggered.[20] Diverse polymeric architectures have been obtained by the crystallization-driven self-assembly (CDSA) approach. Arno *et al.* reported recently the preparation of biocompatible and biodegradable 1D cylindrical and 2D platelet micelles of PCL-*b*-PMMA-*b*-PDMA (CL = ε-caprolactone, MMA = methyl methacrylate) *via* CDSA. Interestingly, they were able to control the dimensions and dispersity of the self-assembled nanostructures.[21] Li *et al.* have reported a poly(L-lactide)-based diblock glycopolymer that assembled into 1D cylinders and 2D diamond-shaped platelets.[22] Qiu and Gao *et al.* have reported the preparation of rectangular and hollow structures from polymer blends.[23]

To date, most reported coil-crystalline self-assembled aggregates with control over their size were prepared *via* CDSA, requiring heating and aging steps. The most common crystalline blocks in these assembled structures are PEO (poly(ethylene oxide)),[24] PCL,[21,25] PE (polyethylene),[26] and PFS (poly(ferrocenyldimethylsilane)).[21,27–30] Only a few studies describe the self-assembly in solution of BCPs containing a fluoropolymer block,[10,12,18,31] and even fewer deal with the CDSA behaviour of these fluorinated BCPs.[10,17]

Dimethyl carbonate (DMC) was shown to be the solvent of choice for the RAFT polymerization of VDF as it allows relatively fast polymerization and an acceptable level of transfer reactions.[3,6] However, one major limitation of the chain extension with the PVDF approach to prepare amphiphilic PVDF-based BCPs is the rather low solubility of hydrophilic macro-CTAs in DMC.

PNIPAM is hydrophilic and soluble in DMC at the required temperature for VDF polymerization (*ca.* 73 °C). PNIPAM features a near body lower critical solution temperature (LCST = 32 °C) and biocompatibility, and is thus very appealing for biomedical applications.[32–36] So far the reports mentioning both PVDF and PNIPAM only describe polymer blends for the preparation of electrospun fibers or flat membranes, or the grafting of PNIPAM on PVDF membranes.[37] To date, the preparation of PNIPAM-*b*-PVDF block copolymers (BCPs) and the study of their self-assembly in selective solvents have not been reported.

The present study recounts the synthesis *via* RAFT sequential addition of VDF, characterization and self-assembly in solution of amphiphilic PNIPAM-*b*-PVDF diblock copolymers. The self-assembly of these BCPs in water led to a wide range of morphologies such as spherical, crumpled-spherical, lamellar and well-defined 2D lenticular aggregates without the need for thermal annealing. The formation and molecular arrangement of these self-assembled structures are discussed, and the thermoresponsive behaviour of the PNIPAM-containing aggregates in water is also examined.

Experimental section

Materials

All reagents were used as received unless otherwise stated. 1,1-Difluoroethylene (vinylidene fluoride, VDF) was supplied by Arkema (Pierre-Bénite, France). *O*-Ethyl-*S*-(1-methoxycarbonyl) ethyl dithiocarbonate (CTA$_{XA}$) was prepared according to the method described by Liu *et al.*[38] *tert*-Amyl peroxy-2-ethylhexanoate (Trigonox 121, purity 95%) was purchased from AkzoNobel (Chalons-en-Champagne, France). Ethanol (EtOH), dimethyl carbonate (DMC), diethyl ether, toluene, laboratory reagent grade hexane (purity >95%) and *N*-isopropylacrylamide (NIPAM, purity 97%) were purchased from Sigma Aldrich. All deuterated solvents were purchased from Eurisotop. NIPAM was recrystallized twice from hexane/toluene (10/1, v/v).

Measurements

Nuclear magnetic resonance (NMR). The NMR spectra were recorded on a Bruker AV III HD Spectrometer (400 MHz for ^{1}H and 376 MHz for ^{19}F).

Coupling constants and chemical shifts are given in hertz (Hz) and parts per million (ppm), respectively. The experimental conditions for recording ^{1}H and ^{19}F NMR spectra were as follows: flip angle, 30°; acquisition time, 4 s (2 s for ^{19}F NMR); pulse delay, 1 s (2 s for ^{19}F NMR); number of scans, 16; and pulse widths of 9.25 and 11.4 μs for ^{1}H and ^{19}F NMR, respectively.

Size-exclusion chromatography (SEC). Size exclusion chromatograms were recorded using a Triple Detection GPC system from Agilent Technologies with its corresponding Agilent software, dedicated to multi-detector GPC calculation. The system used two ResiPore 3 μm 300 × 7.5 mm columns with DMF as the eluent with a flow rate of 1 mL min^{-1} and toluene as the flow rate marker. The detectors used were a PL0390-06034 capillary viscometer and a 390-LC PL0390-0601 refractive index detector. The entire SEC-HPLC system was thermostated at

35 °C. Low dispersity PMMA standards were used for the calibration. The typical sample concentration was 10 mg mL^{-1}.

Differential scanning calorimetry (DSC). DSC measurements were performed on 2–3 mg samples on a TA Instruments DSC Q20 equipped with an RCS90 cooling system. For all measurements, the following heating/cooling cycle was employed: cooling from 40 °C to −73 °C, isotherm at −73 °C for 5 min, first heating ramp from −73 °C to 200 °C at 10 °C min^{-1}, isotherm at 200 °C for 5 min, cooling stage from 200 °C to −73 °C at 10 °C min^{-1}, isotherm plateau at −73 °C for 1 min, second heating ramp from −73 °C to 200 °C at 10 °C min^{-1}, isotherm at 200 °C for 1 min, and final cooling stage from 200 °C to 40 °C. Calibration of the instrument was performed with noble metals and checked before analysis with an indium sample. Melting points were determined at the maximum of the enthalpy peaks.

Thermogravimetric analysis (TGA). TGA was carried out with a TA Instruments TGA G500 from 20 °C to 800 °C. A heating rate of 10 °C min^{-1} was used under an air atmosphere with a flow rate of 60 mL min^{-1}. A dry sample weight of 3 mg was used.

Dynamic light scattering (DLS). DLS measurements of polymer solutions were carried out on a Malvern ZEN1600 using a quartz cuvette. Refractive indices of solvent mixtures were determined using the following equation:

$$\frac{n_m^2 - 1}{n_m^2 + 2} = y_1 \frac{n_1^2 - 1}{n_1^2 + 2} + y_2 \frac{n_2^2 - 1}{n_2^2 + 2} \qquad (1)$$

where n_1, n_2 and n_m are the refractive indices of solvent 1, solvent 2 and the mixture at a certain temperature, and y_1 and y_2 are the volume fractions of solvent 1 and solvent 2. The viscosities of solvent mixtures were obtained from scientific publications[39,40] and an online resource.[41]

Transmission electron microscopy (TEM). TEM studies were conducted using a JEOL 1400+ instrument equipped with a numerical camera, operating with a 120 kV acceleration voltage at 25 °C. To prepare TEM samples, a drop (10.0 µL) of micellar solution was placed onto a Formvar/carbon coated copper grid for 60 s, blotted with filter paper and dried under ambient conditions. All TEM grids were prepared from self-assembly solutions without further dilution.

Scanning electron microscopy (SEM). SEM analyses were conducted using a Hitachi S-4500 instrument operating at a spatial resolution of 1.50 nm at 15 kV energy. The samples were folded on a 45° SEM mount after being coated with an ultrathin layer of electrically conducting platinum deposited by high-vacuum evaporation.

Atomic force microscopy (AFM). AFM images were obtained using a Nanoman V from Bruker Instrument employing the repulsive-tapping mode and a tip (Nanosensors PPP NCL, frequency = 158 kHz, Q factor = 350, $k \sim 14$ N m^{-1}, $R_{tip} \sim 5$ nm).

X-Ray diffraction (XRD). XRD powder patterns were obtained on a Philips X'pert Pro MPD diffractometer using Ni-filtered CuKα1 radiation ($\lambda = 1.5406$ Å) in the Bragg–Brentano scanning mode with a 2θ angle range from 5–60°, and a time per step of 50 s.

Synthesis

PNIPAM-XA synthesis. A typical polymerization (entry 1 in Table 1) of NIPAM was performed as follows: NIPAM (6 g, 53.02×10^{-3} mmol), AIBN (25 mg, 0.15×10^{-3} mmol) and CTA$_{XA}$ (315 mg 1.51×10^{-3} mmol) were dissolved in ethanol (9 mL) in a round bottom flask under magnetic stirring at 20 °C (Scheme 1). The solution was degassed by N$_2$ bubbling for 20 min. Then, the flask was placed in an oil bath at 70 °C for 14 h. The reaction was monitored by ^{1}H NMR and was stopped at NIPAM conversion >99%. The polymer was isolated by precipitation in chilled ether and centrifugation and was dried overnight under vacuum at 30 °C.

^{1}H NMR (400 MHz, (CD$_3$)$_2$CO, δ (ppm), Fig. S1†): 0.9–1.28 (m, −NH−CH(CH$_3$)$_2$; −CH(CH$_3$)(C=O)−), 1.28–1.90 (m, −CH$_2$−CH−(C=O)−; CH$_3$−CH$_2$−O−), 1.90–2.50 (m, −CH$_2$−CH−(C=O)−), 3.64 (−O−CH$_3$, (R CTA)), 3.95–4.25 (m, −NH−CH(CH$_3$)$_2$), 4.50–4.76 (O−CH$_2$−CH$_3$ (Z CTA)), 6.50–8.00 (m, −NH−CH (CH$_3$)$_2$).

PNIPAM-*b*-PVDF synthesis. A typical polymerization of PNIPAM-*b*-PVDF block copolymers was performed as follow: the PNIPAM-XA macro-CTA (Table 1, entry 3) (658 mg, 2.34 ×

Table 1 Synthesis and characterization of the PNIPAM macro-CTA and PNIPAM-*b*-PVDF block copolymers prepared by RAFT polymerization of VDF in DMC at 73 °C

Entry	CTA	M	$[M]_0/[CTA]_0$	Reaction time (h), solvent	Yield (%)	DP$_{(NMR)(R)}$	$M_{n(theo)}$[d] (g mol^{-1})	$M_{n(NMR)(R)}$[e] (g mol^{-1})	$M_{n, SEC}$[f] (g mol^{-1})	$Đ$[f]
1	CTA$_{XA}$	NIPAM	25	14, EtOH	>99	25[a]	3000	3000	3200	1.30
2	PNIPAM$_{25}$-XA	VDF	50	20, DMC	60[b]	35[c]	4900	5300	6700	1.38
3	CTA$_{XA}$	NIPAM	35	14, EtOH	>99	35[a]	4100	4200	4400	1.19
4	PNIPAM$_{35}$-XA	VDF	100	20, DMC	60[b]	60[c]	8000	8000	7500	1.29
5	PNIPAM$_{35}$-XA	VDF	150	20, DMC	61[b]	100[c]	10 000	10 600	9200	1.36
6	PNIPAM$_{35}$-XA	VDF	200	20, DMC	62[b]	150[c]	12 100	13 800	9800	1.43
7	PNIPAM$_{35}$-XA	VDF	600	20, DMC	61[b]	450[c]	27 600	33 000	25 500	1.50

Reactions conditions: (i) (Entries 1 and 3) NIPAM homopolymerization: [I]/[CTA$_{XA}$] = 0.1 with I = AIBN and CTA$_{XA}$ = *O*-ethyl-*S*-(1-methoxycarbonyl) ethyl dithiocarbonate, T = 70 °C; (ii) (entries 2 and 4–7) chain extension of PNIPAM$_{35}$-XA : [I]/[CTA$_{XA}$] = 0.2 with I = Trigonox 121, T = 73 °C. [a] Determined by ^{1}H NMR using eqn (2).† [b] Determined gravimetrically. [e] Determined by ^{1}H NMR using eqn (5). [d] Calculated using the yield as the conversion and eqn (3)† for PNIPAM and (7) for the BCP. [e] Calculated from DP$_{NMR}$ using eqn (4)† for PNIPAM and (6) for PVDF. [f] Determined by SEC (RI detector).
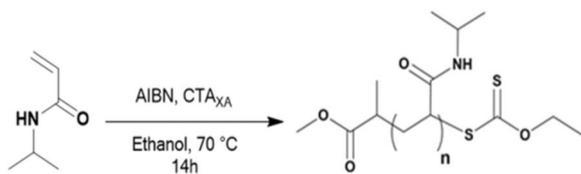
Scheme 1 RAFT/MADIX synthesis of the PNIPAM macro-CTA.
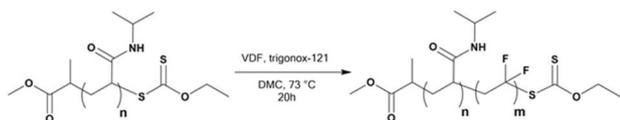
Scheme 2 RAFT/MADIX synthesis of PNIPAM-*b*-PVDF block copolymers using the PNIPAM macro-CTA.

10^{-4} mmol), DMC (5 mL) and the initiator (Trigonox 121) (11 mg, 4.68×10^{-5} mmol) contained in a thick 8 mL Carius tube were sonicated for 5 min or until complete dissolution of PNIPAM-XA (Scheme 2). Then, the tubes were degassed with three freeze–pump–thaw cycles to remove oxygen. The gaseous VDF monomer (1.5 g, 23.42×10^{-3} mmol) was transferred into the Carius tube and condensed in the tube using a liquid nitrogen bath. The tubes were then sealed, before being placed horizontally in a shaking water bath thermostated at 73 °C (see Scheme S1†). After 20 hours, the tube was placed in a liquid nitrogen bath and opened. After cooling to room temperature, the crude sample was precipitated twice in a tenfold excess of chilled pentane. The PNIPAM-*b*-PVDF block copolymers were recovered by centrifugation at 4000 rpm for 5 min in 50 mL conical centrifuge tubes. The polymers were dried overnight under vacuum at 25 °C. Polymerization yields were determined gravimetrically (mass of dried precipitated polymers/mass of monomer introduced in the Carius tube).

^{1}H NMR (400 MHz, $(CD_3)_2CO$, δ (ppm), Fig. S2†): 0.90–1.26 (m, –NH–CH(**CH$_3$**)$_2$–), 1.28–1.90 (m, –**CH$_2$**–CH–NIPAM), 1.90–2.50 (m, –CH$_2$–**CH**–NIPAM), 2.16–2.37 (t, –CF$_2$–**CH$_2$**–CH$_2$–CF$_2$–, VDF–VDF HH reverse addition), 2.66–3.01 (t, –CF$_2$–**CH$_2$**–CF$_2$–, VDF–VDF HT regular addition), 3.8 (s, **CH$_3$**–O–(C=O)–(CH$_3$)CH–), 3.95–4.25 (m, –NH–**CH**(CH$_3$)$_2$), 4.40 (t, CH$_2$–CF$_2$–CF$_2$–**CH$_2$**–S–), 4.60–4.78 (q, –S(C=S)O**CH$_2$**–CH$_3$, $^3J_{HH}$ = 7.1 Hz), 6.09–6.50 (tt, –CH$_2$–CF$_2$–**H**, $^2J_{HF}$ = 55.6 Hz, $^3J_{HH}$ = 4.7 Hz), 6.50–8.00 (m, –**NH**–CH(CH$_3$)$_2$).

^{19}F NMR (376 MHz, $(CD_3)_2CO$, δ (ppm), Fig. S3†): –115.64 (–CH$_2$–CF$_2$–**CF$_2$**–CH$_2$–CH$_2$–, VDF–VDF HH reverse addition), –114.45 (–CH$_2$–**CF$_2$**–H), –113.36 (–CH$_2$–**CF$_2$**–CF$_2$–CH$_2$–CH$_2$–, HH reverse addition), –113.09 (CH$_2$–CF$_2$–**CF$_2$**–CH$_2$–S–), –112.69 (–CH$_2$–**CF$_2$**–CF$_2$–CH$_2$–S–), –107.40 (–C**F$_2$**–CH$_3$), –94.81 (–CH$_2$–CH$_2$–**CF$_2$**–CH$_2$–, TT reverse addition), –93.00 (CH$_3$–O–(C=O)–O–CH$_2$–CH$_2$–**CF$_2$**–, DMC-initiated PVDF), –92.50 (PNIPAM–CH$_2$–**CF$_2$**–), 92.06 (–CH$_2$–**CF$_2$**–CH$_2$–CF$_2$H), –91.43 (–CH$_2$–CH$_2$–CF$_2$–CH$_2$–**CF$_2$**–CH$_2$–CF$_2$–, regular VDF–VDF HT addition), –91.00 (–CH$_2$–**CF$_2$**–CH$_2$–, regular VDF–VDF HT addition).

The degree of polymerization (DP) of the PNIPAM block was calculated from the ^{1}H NMR spectrum of the purified BCP using the equation provided in the ESI (see S2, eqn (2)†). The DP values were then used for the determination of the experimental M_n (S2, eqn (4)†). The degree of polymerization (DP) of the PVDF block was determined using eqn (5) from the integral of the signals (at 0.9–1.28 ppm) corresponding to the methyl groups (–CH$_3$) of the NIPAM units, used as a reference, and the integral of the signals of the –CH$_2$– group of the normal (HT) VDF additions (at 2.70–3.19 ppm). The signal of the –CH$_3$ group of the NIPAM unit is the only signal visible for the BCP with a DP higher than 150. Regarding the reverse (TT) VDF additions, the average number of monomer additions occurring per chain between two degenerative transfers increases with increasing $[VDF]_0/[CTA]_0$. However, the total amount of TT (or HH) VDF additions (intra-chain + chain-end) stabilizes to an identical proportion (*ca.* 4.1%) for PVDF homopolymerization as previously reported by our group.[7] As the signals assigned to these inversions overlap with the signals of the PNIPAM macro-CTA, a 1.041 multiplying factor was employed for the determination of the DP of PVDF.

$$DP_{PVDF} = \frac{\frac{1}{2}\int_{2.70}^{3.19} -\mathbf{CH_2}(HT\,VDF\,additions)}{\frac{1}{6}\int_{0.9}^{1.28} -NH-CH(\mathbf{CH_3})_2} \times 1.041 \quad (5)$$

Molar masses were then calculated using eqn (6) (with $M_{n,\,PNIPAM-XA}$ calculated using eqn (4) and (2)†):

$$M_{n,NMR} = M_{n,PNIPAM-XA} + DP_{PVDF} \times M_{n,VDF} \quad (6)$$

Theoretical molar masses were calculated using eqn (6) with yield = conversion and the $[VDF]_0/[PNIPAM-XA]_0$ ratios listed in Table 1.

$$M_{n,\,theo} = \frac{[VDF]_0}{[PNIPAM-XA]_0} \times Yield \times M_{n,\,VDF} + M_{n,\,PNIPAAm-XA} \quad (7)$$

where $M_{n,\,VDF} = 64.03$ g mol^{-1}.

Self-assembly

Preparation of block copolymer solutions. Stock solutions of 2 mg mL^{-1} of block copolymer were prepared in DMF, acetone or THF at room temperature using magnetic stirring until complete solubilisation.

Nanoprecipitation. Glass vials containing 2 mL of a non-solvent and a magnetic bar were placed on a stirring plate. To each vial, 0.1 mL of block copolymer solution (2 mg mL^{-1}) in DMF was added dropwise. After 1 h of stirring, the samples were analysed by DLS and TEM. The final concentration of the block copolymer was 0.1 mg mL^{-1} in DMF : water (1 : 20).

Note: The difference between precipitation and nanoprecipitation is that polymer precipitation refers to a polymer purification technique in which a solution of the polymer is cast dropwise into a non-solvent. Upon contact between the two miscible solvents, the polymer crashes out of the solution while the impurities remain in the solvent phase. In the case of block copolymers, for an efficient precipitation, the different blocks must be insoluble in the non-solvent. In con-

trast, polymer nanoprecipitation refers to a BCP self-assembly technique in which a polymer solution is cast dropwise into a solvent selective for one of the blocks. The insoluble block forms the core of the self-assembled structure while the soluble block acts as a steric stabilizer of the polymer morphology.

Solvent switch. Vials containing 0.5 mL of the stock solution (2 mg mL^{-1}) in different solvents (THF, DMF and acetone) were placed on a stirring plate. Water (2, 3 or 4 mL) was added dropwise using a syringe pump at a fixed rate of 4 mL h^{-1}. 10 μL was used to prepare TEM samples at 1 : 4, 1 : 6 and 1 : 8 solvent : non-solvent ratios.

Thermally-induced crystallisation-driven self-assembly (TI-CDSA). A 5 mg mL^{-1} PNIPAM$_{25}$-b-PVDF$_{35}$ solution in DMF was self-assembled by adding water to a 1 : 1 solvent : non-solvent mixture, and this solution was then heated at 90 °C for 30 min and slowly cooled down to room temperature.

Results and discussion

PNIPAM-XA macro-CTAs were synthesized by RAFT polymerization using CTA$_{XA}$ following protocols described previously by Sistach *et al.*[42] The reactions were stopped when the NIPAM conversion reached at least 99%. After purification by precipitation, ^{1}H NMR spectroscopy of the resulting PNIPAM-XA macro-CTAs was performed to determine their molar masses and degrees of polymerization (DPs). Amphiphilic PVDF-based block copolymers with PVDF of different DPs were prepared from these PNIPAM macro-CTAs (see Table 1) by chain extension with VDF in DMC (a common solvent for PNIPAM and PVDF allowing a relatively high VDF polymerization rate) using Trigonox 121 as the radical initiator.

^{19}F NMR spectroscopy of the resulting polymers showed the successful chain extension of PVDF from the PNIPAM macro-CTAs. The presence of the signals of –C**F$_2$**–C**F$_2$**–CH$_2$–XA at δ = –113.09 ppm and –112.69 ppm and of –C**H$_2$**–XA at δ = 4.45 ppm and –C**H$_2$**–SC(S)–O–C**H$_2$**– at δ = 4.72 ppm in the ^{19}F NMR and ^{1}H NMR spectra, respectively (Fig. S4 and S3b†), indicates the formation of the diblock copolymers.

As expected, the polymerization of VDF was accompanied by a non-negligible amount of transfer to DMC. The characteristic signals of these transfer reactions can be observed in the ^{1}H NMR spectrum (Fig. S3a and S3b†) as a triplet of triplets at 6.3 ppm corresponding to the –CF$_2$**H** chain end, and in the ^{19}F NMR spectrum (Fig. S4†) as a doublet of triplet at –114.45 ppm (–CF$_2$**H**) and a multiplet at –107.3 ppm corresponding to the C**F$_2$**CH$_3$ chain end. The ^{1}H NMR spectrum also shows a singlet at 5.77 ppm assigned to the DMC–xanthate adduct (CH$_3$O(C=O)O–CH$_2$–XA, Fig. S3a†), eliminated by the precipitation step (Fig. S3b†), and a well-defined triplet at 4.35 ppm and a singlet at 3.73 ppm assigned to the –CH$_2$– and –CH$_3$ groups of the DMC moieties of the DMC-initiated PVDF chains, respectively. The RAFT polymerization of VDF is accompanied by a progressive loss of the chain-end functionality (loss of the xanthate group).[6,7] This phenomenon has pre-

viously been reported for the synthesis of PVAc-b-PVDF BCPs.[17] This loss of functionality can be very extensive. Here the remaining molar fraction of the xanthate-functionalized chain end derived from the ^{19}F NMR spectra (Fig. S4 and calculations S5†) was 19.2, 5.4, 2.3 and 1.4% for the PNIPAM$_{35}$-b-PVDF$_{60}$, PNIPAM$_{35}$-b-PVDF$_{100}$, PNIPAM$_{35}$-b-PVDF$_{150}$ and PNIPAM$_{35}$-b-PVDF$_{450}$ copolymers respectively.

Despite these transfer reactions and loss of functionality, the chain extension of PNIPAM macro-CTAs with PVDF produced relatively well-defined BCPs with dispersity below 1.50 and monomodal SEC traces without shoulders or significant tailing (Fig. 1). These SEC traces also show a clear shift towards higher molar masses with increasing DP of PVDF.

These amphiphilic block copolymers were then used to prepare self-assembled morphologies in different solvents.

Note on solvents: in the following self-assembly study, three solvents were used as good solvents for both the PNIPAM and PVDF segments: acetone, THF and DMF. All of these solvents are good solvents of PNIPAM. Bottino *et al.*[43] investigated the solubility of 450 000 g mol^{-1} PVDF in 46 liquids and concluded that DMF is a solvent of PVDF, while acetone and THF are good swelling agents. The PVDF blocks synthesized here have a much lower molar mass (PVDF with a DP = 450 would have an M_n value of 28 800 g mol^{-1}); it is thus not surprising that THF and acetone behave as good solvents especially considering the low concentrations used (2 mg mL^{-1}, whereas Bottino *et al.* aimed at preparing 100 mg mL^{-1} solutions). In addition, the self-assembly described and discussed below concerns PNIPAM-b-PVDF block copolymers whose solubility would be enhanced by the PNIPAM block. These copolymers indeed readily dissolved in acetone and in THF, although gentle heating of the solution was necessary for the latter.

The self-assembly of amphiphilic BCPs using rapid solvent exchange usually leads to the formation of colloidal objects *via* microphase separation. The final structure of these block
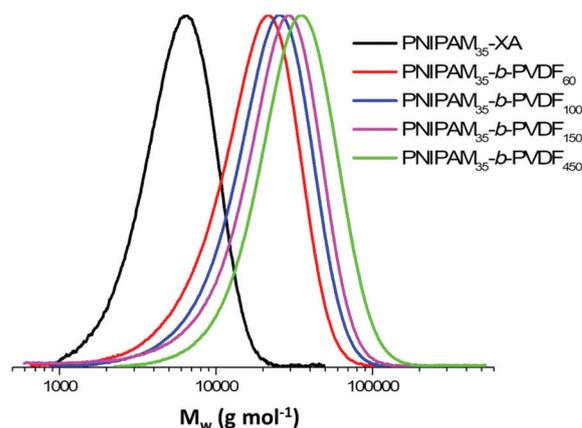

Fig. 1 Normalized SEC traces (viscometric detector) of: PNIPAM$_{35}$-XA (black trace), PNIPAM$_{35}$-b-PVDF$_{60}$ (red trace), PNIPAM$_{35}$-b-PVDF$_{100}$ (blue trace), PNIPAM35-b-PVDF150 (pink trace) and PNIPAM35-b-PVDF450 (green trace) after purification by precipitation in chilled ether for PNIPAM or cold pentane for the BCPs.

copolymer colloids is primarily dictated by the volume fraction of the blocks and by the interfacial surface tensions. A feature of the nanoprecipitation process (which consists of the drop-wise addition of the BCP solution into a selective solvent under vigorous stirring) is the ability to access kinetically trapped morphologies in nonequilibrium states due to short mixing time. This kinetic trapping is even more pronounced in the case of semicrystalline polymers such as PVDF. In such a case, the PVDF segments often crystallize before the polymer chains can reach the equilibrium morphology during phase separation.

The self-assembly of the PNIPAM-b-PVDF block copolymers was achieved *via* nanoprecipitation in water from a dilute DMF BCP solution. After stirring for 1 h, the samples were analyzed by DLS and TEM. This approach led to the formation of spherical aggregates of roughly 20 to 60 nm diameter (Fig. 2).

The particle sizes measured by TEM were in good agreement with the average hydrodynamic diameters measured by DLS (Fig. S6†) ranging between 30 and 80 nm and increasing with the DP of the core-forming PVDF block.

Solvent switch experiments (slow addition of water into a solution of PNIPAM$_{35}$-b-PVDF$_X$ in DMF, THF or acetone) were also performed using three (1 : 4, 1 : 6 and 1 : 8) solvent : non-solvent ratios.

When DMF was employed as a common solvent and water as a selective solvent for the PNIPAM block, irregular crumpled spherical aggregates were obtained with all the solvent : non-solvent ratios tested (Fig. 3). CDSA self-assembled polymers
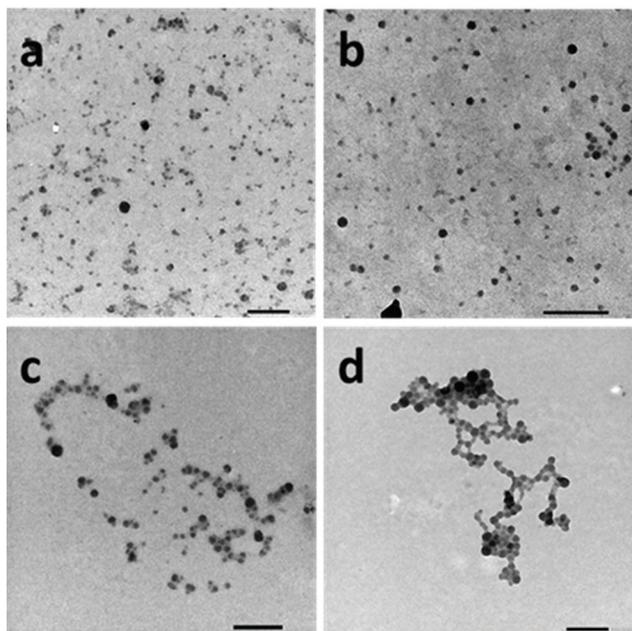

Fig. 3 TEM images of crumpled aggregates obtained by the solvent switch of (a and b) PNIPAM$_{25}$-b-PVDF$_{35}$ and (b and c) PNIPAM$_{35}$-b-PVDF$_{450}$ in DMF at 2 mg mL^{-1}. Final concentration = 0.4 mg mL^{-1} in the DMF : water (1 : 4) mixture.

are thought to favor the formation of aggregates with low inter-facial curvature.[44] These morphologies may be due to the co-nonsolvency of PNIPAM blocks when the DMF molar fraction changes in the solvent mixture, provoking the collapse of the metastable aggregates before attaining the spherical mor-phology obtained by nanoprecipitation. A reversible collapse–swelling behavior has already been observed in PNIPAM latexes in water–DMF mixtures.[45] When the DMF molar frac-tion was between 0.02 and 0.25, the latex volume was smaller than that observed in pure water.

When THF was used as the good solvent for both blocks, solvent switch protocols led to the formation of flat sheet mor-phologies, presumably 2D bilayer aggregates (Fig. 4).

When acetone was used as the good solvent, well-defined flat 2D lenticular morphologies with relatively low dispersity and good dimensional control were obtained (Fig. 5–7).

In the case of the PNIPAM$_{25}$-b-PVDF$_{35}$ BCP, these lenticular objects had an average length and width of 600 nm and 250 nm, respectively, and ragged edges that seemed to be formed by aggregation of smaller 1D rod-like aggregates (Fig. 5b and S7†). Crystallization-induced aggregation of micelles has already been observed for PB-b-PEO BCPs.[46] These short 1D micelles have an average length and width of 98 and 19 nm, respectively.

In contrast, the PNIPAM-b-PVDF BCP prepared from the PNIPAM$_{35}$ macro-CTA self-assembled into lenticular objects with much smoother-looking edges with average lengths of 200, 600, 1000 and 2300 nm and average widths of 90, 250, 400 and 850 nm for PVDF DPs of 60, 100, 150 and 450, respectively (Fig. 6).

Fig. 2 TEM images of spherical aggregates prepared from (a) PNIPAM$_{35}$-b-PVDF$_{60}$, (b) PNIPAM$_{35}$-b-PVDF$_{100}$, (c) PNIPAM$_{35}$-b-PVDF$_{150}$ and (d) PNIPAM$_{35}$-b-PVDF$_{450}$. All samples were prepared by nanopreci-pitation in water from BCP solutions in DMF at 2 mg mL^{-1} (final concen-tration = 0.1 mg mL^{-1} in DMF : water (1 : 20)). Scale bars correspond to 200 nm.
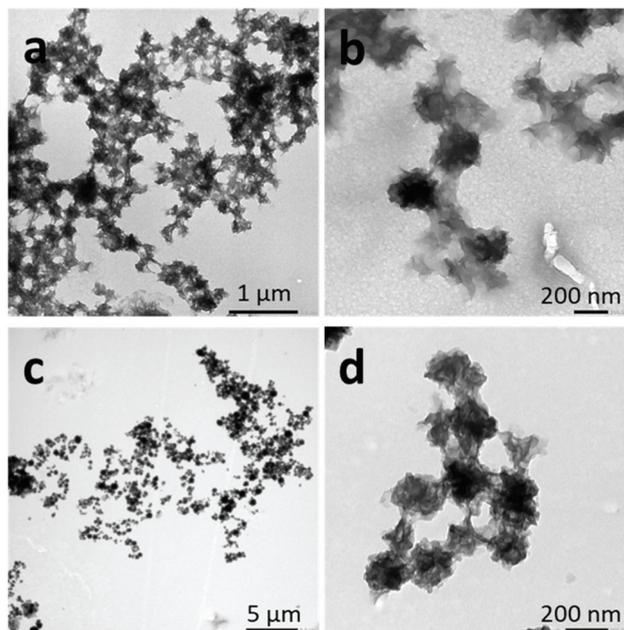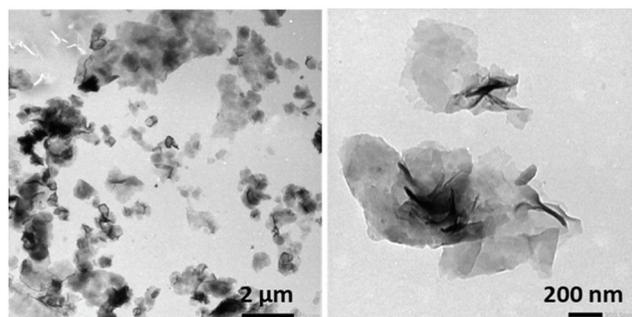
Fig. 4 TEM images of the bilayer aggregates obtained by the solvent switch of PNIPAM$_{35}$-b-PVDF$_{450}$. Initial polymer concentration in THF = 2 mg mL^{-1}. Final concentration = 0.4 mg mL^{-1} in the THF : water (1 : 4) mixture. The water addition rate was 4 mL h^{-1}.
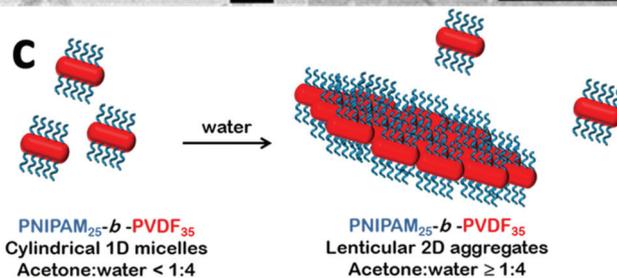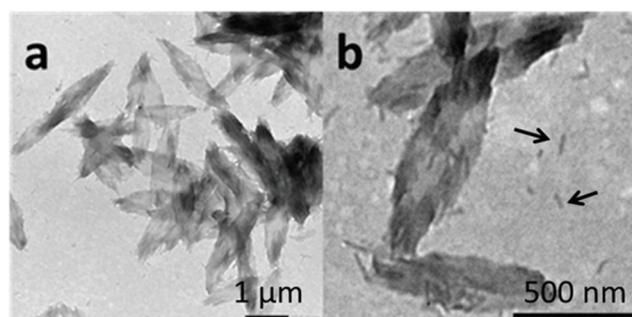

Fig. 5 TEM images of 2D lenticular micelles obtained by self-assembly *via* the solvent switch protocol of PNIPAM$_{25}$-b-PVDF$_{35}$ (a and b). Initial BCP concentration in acetone = 2 mg mL^{-1}. Final concentration = 0.4 mg mL^{-1} in the acetone : water (1 : 4) mixture. The non-solvent addition rate was 4 mL h^{-1}. The arrows indicate the isolated 1D micelles. Schematic representation of the proposed mechanism of the formation of PNIPAM$_{25}$-b-PVDF$_{35}$ lenticular aggregates from short 1D rod-like micelles (c).

The length and width of the lenticular 2D micelles showed a linear dependence on the degree of polymerization of PVDF up to 150 (Fig. S7†). The loss of linearity observed for the PVDF$_{450}$ BCP may be due to the poor accuracy of the DP value calculated from weak NMR signals. The real DP of this PVDF block might be closer to 300 than the calculated 450.

Due to their relatively large size (in the micrometer range), these morphologies were not colloidally stable and sedimented over time (*ca.* 14 h). However, they were easily redispersed by simple shaking.

The shape and size of these morphologies were not affected by the concentration of the initial BCP solutions over the con-
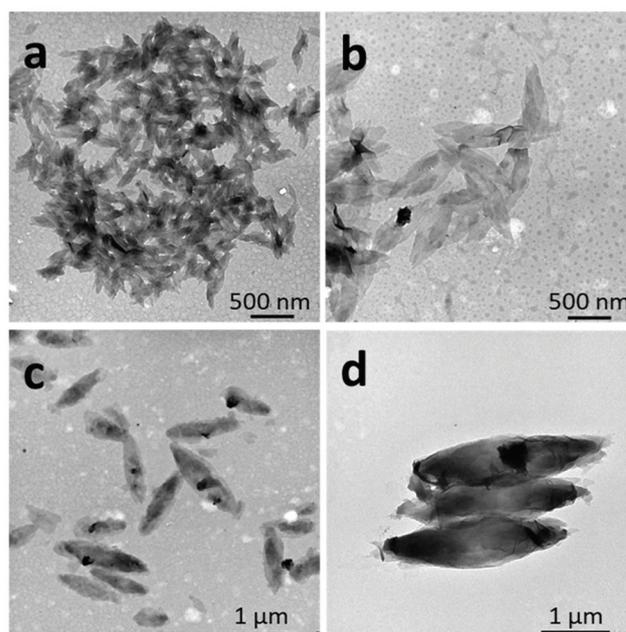

Fig. 6 TEM images of representative self-assembled 2D lenticular morphologies prepared from (a) PNIPAM$_{35}$-b-PVDF$_{60}$, (b) PNIPAM$_{35}$-b-PVDF$_{100}$, (c) PNIPAM$_{35}$-b-PVDF$_{150}$ and (d) PNIPAM$_{35}$-b-PVDF$_{450}$. All samples were prepared *via* the solvent switch in acetone : water mixtures. Initial polymer concentration in acetone = 2 mg mL^{-1}. Final concentration = 0.4 mg mL^{-1} in the acetone : water (1 : 4) mixture. Water addition rate = 4 mL h^{-1}.
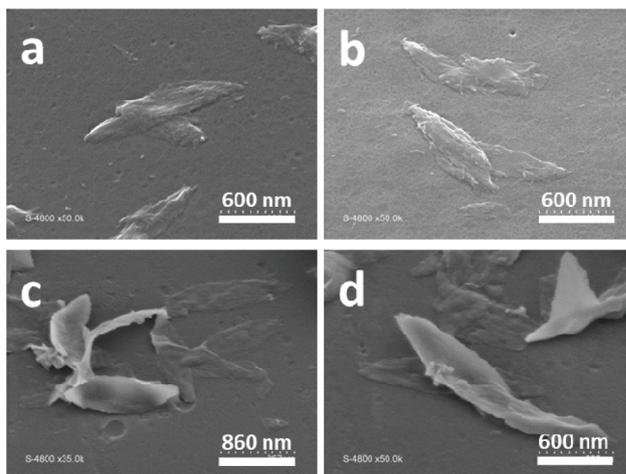

Fig. 7 SEM images of 2D lenticular objects prepared from: PNIPAM$_{25}$-b-PVDF$_{35}$ (a and b) and PNIPAM$_{35}$-b-PVDF$_{450}$ (c and d).

centration range studied (Fig. S9†). In addition, once the self-assembled structures were formed in an acetone : water mixture (1 : 4), they did not evolve within the 2-week observation time; they thus seem to be kinetically trapped. Increasing the water content of the self-assembly media or decreasing the acetone content (by evaporation) of the solvent mixture only affected the aggregation rate. However, the lenti-

cular shape of these assemblies was strongly affected by the ratio of the length of the hydrophobic and hydrophilic blocks. The morphologies were rather patchy with sharp edges for PNIPAM$_{25}$-b-PVDF$_{35}$ (Fig. 5), while better-defined objects were found for PNIPAM$_{35}$-b-PVDF$_n$ BCPs with $n > 60$ (Fig. 6). Better-defined lenticular objects were obtained as higher DPs of PVDF were targeted.

The PVDF fraction of these BCPs was crystalline as determined from DSC thermograms (Fig. S10†). However, the PVDF crystallinity could not be observed on these self-assembled structures by electron diffraction during TEM analysis. This was ascribed to the thinness of these lenticular objects, undergoing fast amorphisation under the electron beam.

Nevertheless, the XRD diffraction pattern (Fig. 8) obtained from freeze-dried 2D aggregates formed via self-assembly of a 2 mg mL^{-1} solution of PNIPAM$_{35}$-b-PVDF$_{100}$ BCPs in acetone : water (1 : 4) suggests that the PVDF phase of these aggregates was indeed crystalline and in the α-form with characteristic 2θ peaks at 17.6 (100), 18.5 (020), 19.8 (110), 26.5 (021), 36.1 (200) and 39 (002).[47] The α-crystal phase is hexagonal, with aligned polymer chains anti-parallel to each other, in the conformation of *trans-gauche–trans-gauche'* (TGTG'). The α-form is the most common and spontaneously forming crystalline form of PVDF.[48,49]

SEM images of the lenticular objects (Fig. 7) suggest that they can fold and twist to an extent, and are often stacked on top of each other, making their characterization quite difficult. AFM images (Fig. 9) confirmed the stacking of the objects but also revealed the increase of their thickness with the increase of the DP of the PVDF block.

The presence of short cylindrical 1D micelles (Fig. 6b) suggests that the lenticular morphologies may form by aggregation of these smaller structures. In addition, the thickness of the lenticular aggregates seems to be related to the length and folding of the PVDF segments. As the degree of polymerization of the PVDF core-forming block increased, the thickness of the aggregates increased.
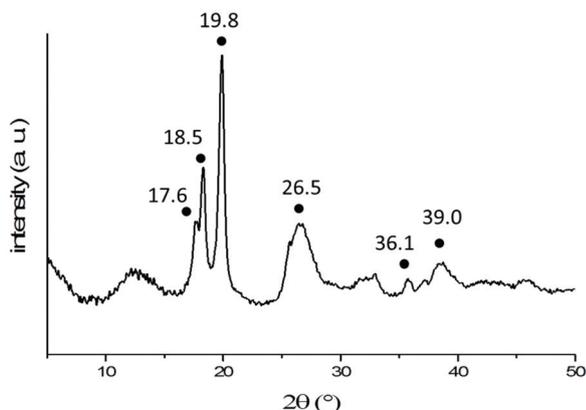

Fig. 8 X-Ray diffraction pattern of freeze-dried 2D lenticular aggregates obtained from PNIPAM$_{35}$-b-PVDF$_{100}$ BCPs. The black dots indicate the characteristic peaks of the α-phase of PVDF.

Since the PVDF in these aggregates is crystallized in the α-form, the number of foldings of the PVDF chains can be estimated. In the α-form crystal, 0.462 nm is the dimension of 2.5 VDF units in that specific conformation.[50] Crystallized PVDF chains of DP 60, 100, 150 and 450 would thus extend over 11.1, 18.5, 27.7 and 83.2 nm, respectively, in the α-form. The thicknesses of the lenticular objects derived from the AFM topographic images shown in Fig. 9 are in agreement with these calculations for PVDF$_{60}$, PVDF$_{100}$ and PVDF$_{150}$. Fig. 9a, corresponding to PNIPAM$_{35}$-b-PVDF$_{60}$, shows a thickness of 10–15 nm. Fig. 9b and Fig. S11,† corresponding to PNIPAM$_{35}$-b-PVDF$_{100}$, shows a thickness of about 40 nm. This value is roughly twice the calculated length of PVDF$_{100}$ (18.5 nm). This is because the AFM images in Fig. 9b show two aggregates stacked on top of each other. The thickness measured for PNIPAM$_{35}$-b-PVDF$_{150}$ in Fig. 9c is 30 nm which also is in agreement with the corresponding calculated value of 27.7 nm. In the case of the PNIPAM$_{35}$-b-PVDF$_{450}$ BCP, the calculated thickness of 83.2 nm does not match the thickness of 10 nm measured in Fig. 9d. This discrepancy may be explained by considering that in this case the crystallized PVDF chains are folded rather than extended. Indeed, the calculated and measured values can be reconciled if the PVDF$_{450}$ chains were folded 7 times (or 5 times if the real DP is actually 300 as mentioned above). This explanation is also consistent with the observation of larger aggregates for a higher PVDF degree of polymerization.

As PVDF chains fold more, the repulsion between the PNIPAM hydrophilic chains decreases, allowing the formation of larger but thinner 2D aggregates.

Since the degree of crystallinity can be modified via temperature annealing, a heating and cooling treatment was applied to the crumpled structures (Fig. 3) obtained by self-assembly from the DMF solution. In addition, at higher temperature, the solubility of PVDF in a DMF/water mixture increases. A 5 mg mL^{-1} PNIPAM$_{25}$-b-PVDF$_{35}$ BCP solution in DMF was self-assembled by adding water to reach a 1 : 1 solvent : non-solvent ratio; then this solution was heated at 90 °C for 30 min and slowly cooled down to room temperature. Fig. S12† shows that this temperature treatment led to a mixture of ill-defined aggregates and spindle-shaped morphologies with lengths ranging from 300 nm to 1 μm. The crumpled sphere to spindle morphology transition is thought to proceed via temperature-induced crystallization-driven self-assembly (TI-CDSA) due to the degradation of the solvent quality for the crystallisable block caused by the slow cooling.[20]

To sum up, five different morphologies were observed by the self-assembly of these PNIPAM-b-PVDF BCPs as presented in Table 2.

Without surprise, the solvents used and the self-assembly protocol played a crucial role in the resulting BCP structures. The same DMF/water binary mixture of solvents afforded three distinct morphologies depending on the self-assembly process used. Spherical nanoparticles were obtained when nanoprecipitation was used while a solvent switch protocol led to the formation of crumpled spheres. These crumpled spheres evolved
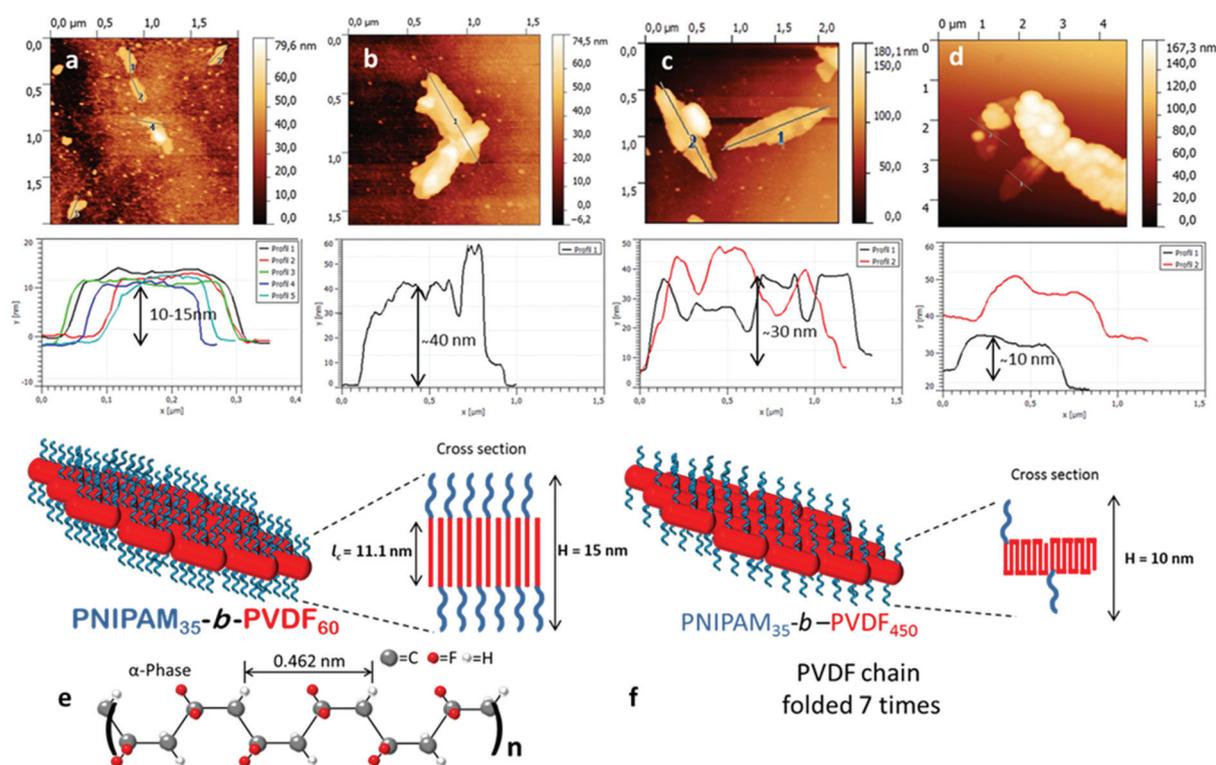
Fig. 9 AFM images of 2D lenticular aggregates prepared from: (a) PNIPAM$_{35}$-b-PVDF$_{60}$, (b) PNIPAM$_{35}$-b-PVDF$_{100}$, (c) PNIPAM$_{35}$-b-PVDF$_{150}$, and (d) PNIPAM$_{35}$-b-PVDF$_{450}$. (e and f) Schematic representation and suggested molecular arrangement of the 2D lenticular aggregates formed by the assembly of PNIPAM$_{35}$-b-PVDF$_{60}$ and PNIPAM$_{35}$-b-PVDF$_{450}$ respectively.

Table 2 Shape of self-assembled aggregates for the different polymer systems and protocols

BCP	NP DMF : H$_2$O	SS DMF : H$_2$O	SS THF : H$_2$O	SS Acetone : H$_2$O	SS + TI-CDSA DMF : H$_2$O
PNIPAM$_{25}$-b-PVDF$_{35}$	Spherical	Crumpled spherical	Flat sheet	Lenticular^a	Spindle
PNIPAM$_{35}$-b-PVDF$_{60}$	Spherical	—	—	Lenticular	—
PNIPAM$_{35}$-b-PVDF$_{100}$	Spherical	—	—	Lenticular	—
PNIPAM$_{35}$-b-PVDF$_{150}$	Spherical	—	—	Lenticular	—
PNIPAM$_{35}$-b-PVDF$_{450}$	Spherical	Crumpled spherical	Flat sheet	Lenticular	—

NP, SS and TI-CDSA stand for nanoprecipitation, solvent switch and temperature-induced crystallization-driven self-assembly protocols respectively. ^{a} Sharp edges. In all PNIPAM$_{35}$-b-PVDF$_m$ systems examined, the size of the nanoaggregates increased with increasing DP of the PVDF block. The initial concentration for nanoprecipitation and solvent switch protocols was 2 mg mL^{-1} (5 mg mL^{-1} in the case of the sample used for TI-CDSA). The rate of selective solvent addition was 4 mL h^{-1} for the solvent switch protocol and the solvent was added dropwise during nanoprecipitation. The TI-CDSA sample was placed for 30 min in an oil bath at 90 °C and slowly cooled after that time.

into spindle-like aggregates when a TI-CDSA protocol was applied. The solvent switch protocol produced 2D lamellar aggregates in the case of THF/water and well-defined lenticular shaped aggregates of increasing length and width with increasing DP of the PVDF block in an acetone/water binary mixture.

Thermoresponse of PNIPAM-b-PVDF lenticular nanoparticles

The thermoresponsiveness of the PNIPAM-b-PVDF lenticular aggregates was also investigated. The usual turbidity test could not be used due to the poor colloidal stability of the BCP aggregates even at room temperature. Since the LCST of PNIPAM is affected by the presence of organic solvents, acetone was com-

pletely removed from the PNIPAM-b-PVDF BCP suspensions under reduced pressure using a rotary evaporator at room temperature to prepare an aqueous suspension of the self-assembled morphologies. TEM analysis of this suspension (Fig. S13†) showed that the size and shape of the assemblies were not affected by the removal of acetone.

The samples were heated at 50 °C under gentle stirring for a period of 30 min and then cooled down to room temperature.

Fig. 10 shows the effect of temperature on the aqueous PNIPAM-b-PVDF BCP morphologies. The lenticular objects partly lost their well-defined shape, crumpled to an extent and also broke into smaller flat sheet aggregates. The change of
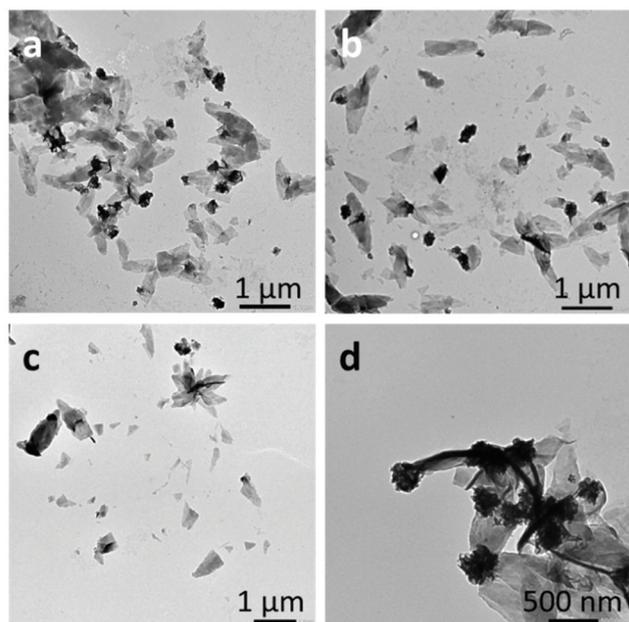
Fig. 10 TEM images of the representative samples of 2D lenticular aggregates prepared from: (a and b) PNIPAM$_{35}$-*b*-PVDF$_{150}$ BCPs and (c and d) PNIPAM$_{35}$-*b*-PVDF$_{450}$ BCPs in pure water after being heated at 45 °C for 30 min.

conformation of the PNIPAM blocks provoked by the decrease of the PNIPAM solubility in water at $T >$ LCST$_{PNIPAM}$ is thought to be responsible for the aggregates to irreversibly break or evolve into the observed crumpled morphologies.

Conclusions

A series of relatively well-defined PNIPAM-*b*-PVDF semicrystalline amphiphilic diblock copolymers were successfully synthesized by RAFT polymerization using PNIPAM macro-CTAs. These diblock copolymers were fully characterized by ^{1}H and ^{19}F NMR spectroscopy, TGA, DSC and SEC. Despite the loss of the xanthate functionality observed when DPs of PVDF higher than 50 were targeted, BCPs with relatively low dispersity ($Đ <$ 1.50) were achieved. Due to their amphiphilic nature, these BCPs had the ability to self-assemble in aqueous solutions. The self-assembled structures were characterized by TEM, SEM and AFM microscopies. The final structures were strongly affected by the choice of common solvents and the self-assembly protocol used. Five types of morphologies were obtained depending on the common solvent and the self-assembly protocol. Surprisingly, self-assembly from acetone solutions led to the formation of well-defined 2D lenticular aggregates whose dimensions increased with the degree of polymerization of PVDF, reaching a length of 2.3 μm for the longest PVDF synthesized. These aggregates were morphologically stable over time and accessible even at higher concentrations. A formation pathway and a possible molecular arrangement of these 2D lenticular aggregates were proposed.

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