



Cite this: *Nanoscale*, 2021, **13**, 9773

Development of a universal conductive platform for anchoring photo- and electroactive proteins using organometallic terpyridine molecular wires†

Margot Jacquet, ^a Miriam Izzo, ^a Silvio Osella, ^b Sylwia Kozdra, ^c Paweł P. Michałowski, ^c Dariusz Gołowicz, ^d Krzysztof Kazimierczuk, ^d Maciej T. Gorzkowski, ^e Adam Lewera, ^e Marian Teodorczyk, ^c Bartosz Trzaskowski, ^b Rafał Jurczakowski, ^f Daniel T. Gryko ^{*g} and Joanna Kargul ^{*a}

The construction of an efficient conductive interface between electrodes and electroactive proteins is a major challenge in the biosensor and bioelectrochemistry fields to achieve the desired nanodevice performance. Concomitantly, metallo-organic terpyridine wires have been extensively studied for their great ability to mediate electron transfer over a long-range distance. In this study, we report a novel stepwise bottom-up approach for assembling bioelectrodes based on a genetically modified model electroactive protein, cytochrome *c*₅₅₃ (cyt *c*₅₅₃) and an organometallic terpyridine (TPY) molecular wire self-assembled monolayer (SAM). Efficient anchoring of the TPY derivative (**TPY-PO(OH)₂**) onto the ITO surface was achieved by optimising solvent composition. Uniform surface coverage with the electroactive protein was achieved by binding the cyt *c*₅₅₃ molecules *via* the C-terminal His₆-tag to the modified TPY macromolecules containing Earth abundant metallic redox centres. Photoelectrochemical characterisation demonstrates the crucial importance of the metal redox centre for the determination of the desired electron transfer properties between cyt and the ITO electrode. Even without the cyt protein, the ITO-TPY nanosystem reported here generates photocurrents whose densities are 2-fold higher than those reported earlier for ITO electrodes functionalised with the photoactive proteins such as photosystem I in the presence of an external mediator, and 30-fold higher than that of the pristine ITO. The universal chemical platform for anchoring and nanostructuring of (photo)electroactive proteins reported in this study provides a major advancement for the construction of efficient (bio)molecular systems requiring a high degree of precise supramolecular organisation as well as efficient charge transfer between (photo)redox-active molecular components and various types of electrode materials.

Received 15th December 2020,
Accepted 23rd April 2021

DOI: 10.1039/d0nr08870f
rsc.li/nanoscale



^aSolar Fuels Laboratory, Centre of New Technologies, University of Warsaw, Banacha 2C, 02-097 Warsaw, Poland. E-mail: j.kargul@cent.uw.edu.pl

^bChemical and Biological Systems Simulation Laboratory, Centre of New Technologies, University of Warsaw, Banacha 2C, 02-097 Warsaw, Poland

^cLukasiewicz Research Network – Institute of Microelectronics and Photonics, Aleja Lotników 32/46, 02-668 Warsaw, Poland

^dLaboratory of NMR Spectroscopy, Centre of New Technologies, University of Warsaw, Banacha 2C, 02-097 Warsaw, Poland

^eCatalysis and Surface Chemistry Laboratory, Centre of Biological and Chemical Sciences, Faculty of Chemistry, University of Warsaw, ul. Żwirki i Wigury 101, 02-089 Warsaw, Poland

^fElectrochemistry of New Materials, Centre of Biological and Chemical Sciences, Faculty of Chemistry, University of Warsaw, ul. Żwirki i Wigury 101, 02-089 Warsaw, Poland

^gInstitute of Organic Chemistry, Polish Academy of Sciences, Kasprzaka 44/52, 01-224 Warsaw, Poland. E-mail: dtgryko@icho.edu.pl

† Electronic supplementary information (ESI) available. See DOI: [10.1039/d0nr08870f](https://doi.org/10.1039/d0nr08870f)

Introduction

Biohybrid technologies provide a promising approach to develop cost-effective, sustainable nanomaterials for a wide range of applications^{1–5} combining evolutionary optimised natural functionalities with advanced nanoengineering.⁶ In this context, metalloproteins play an important role due to their unique redox properties and fast (on ns–μs timescale) electron transfer (ET). One such protein, cytochrome *c* (cyt *c*) has been used as an electroactive component of various types of devices including biosensors and biosolar cells.^{7,8} This small protein containing a redox-active haem group is a crucial electron mediator in the mitochondrial respiratory chain and in photosynthesis. It has been intensively studied as a model system to understand the ET processes occurring in biological systems⁹ as well as in hybrid nanomaterials.¹⁰ Indeed, its ability to relay electrons has found many applications in

various fields ranging from biosensors,¹¹ biocatalysis,¹² molecular memory¹³ and biophotovoltaics.¹⁴ In natural oxygenic photosynthesis, cyt *c* delivers water-derived electrons to the photo-oxidised primary electron donor of photosystem I (PSI), one of the key light-harvesting macromolecular components of the photosynthetic machinery. This feature has been utilised in artificial photosynthetic systems, whereby cyt *c* forms a bio-organic conductive interface between PSI and the electrode surface,^{15,16} at the same time orienting PSI so as to achieve a significant improvement of photocurrent output.^{17,18} A key factor for the optimal performance of such biohybrid devices is to ensure the efficient electronic communication between cyt *c* and the electrode surface, which can be achieved either *via* direct electron transfer (DET) or in a mediated fashion (MET). Depending on the nature of the ET many factors can influence the efficiency of this process such as the energetic compatibility with the electrode material, the surrounding solvent, as well as the mediator itself and its concentration in the case of the mediated ET.¹⁹ A common important factor is the nature of the interface between cyt *c* and the electrode surface that affects ET kinetic parameters and modifies the distance and orientation of the electroactive protein.^{11,20}

In the molecular wiring approach, the design of the molecular wire's structure together with the appropriate orientation of the photo-electroactive protein towards the electrode surface is important to achieve the highest possible charge transport efficiency and minimise wasteful back reactions. The rational approach (concluded both from the experimental data and theoretical calculations) for designing the molecular wire with improved conductivity is based on increasing the conjugation level of the electronic molecular structures.^{21–28} A promising strategy relies on the use of organometallic terpyridine (TPY) wires covalently attached to the conductive transparent material.^{29–31} The geometry of TPY complexes allows for the construction of a well-organised modular architecture that is amenable for the fine tuning of the conductive properties through the incorporation of various metallic redox centres.

It was reported that some self-assembled molecular wires, based on TPY complexes, can be highly conductive even up to ~40 nm in length. This high conductivity was attributed to the presence of coordination redox-active centres in the highly rigid organic core.³² Remarkably, it was shown that there is no strong dependence between electron work function and the molecular length of the TPY wire.³² Another important factor to take into account is the development of the appropriate structure of the highly conductive TPY wire for the domain-specific anchoring of an electroactive protein such as cyt *c*. In this context, the bottom-up strategy based on step-by-step construction of the wire offers an advantage over other approaches as it permits to fine-tune the ET parameters by tuning the molecular structure of the wire.^{33,34} These structural and electronic features altogether make the coordinative TPY ligand-based conductive organic interface an excellent material of choice for efficient ET between redox-active proteins and the compatible electrode, together with its ability of strong and selective binding to protein molecules.

In this work, we report for the first time the rational design and cost-effective production of bio-hybrid nanodevices based on cyt *c*₅₅₃ and well-organised metallo-organic terpyridine wires. Highly π -conjugated TPY ditopic ligands and phosphate derivatives were synthesised and used as the conductive building blocks in the hybrid nanoarchitecture. The optimised step-by-step construction of the nano-assemblies on transparent oxide-based electrodes *via* the bottom up approach and using non-toxic Earth abundant metal redox centres is described. We demonstrate that these molecules form a well-defined surface-coverage SAM that can be used as a universal, highly conductive platform for the domain-specific anchoring of electroactive proteins. The efficient ET amenable for fine-tuning through the incorporation of metallic redox centres within the TPY structure paves the way for the application of this promising novel interface in various types of bionanodevices ranging from solar cells and solar-to-fuel devices to nanosensors.

Experimental

General materials

All chemical reagents and solvents used for synthesis were purchased from commercial sources (Aldrich, Acros and VWR) and used without further purification. ITO glass substrates were purchased from Ossila (15 mm × 20 mm, 1.1 mm thickness). Prior to use, ITO electrodes were cut in 15 × 10 mm for all following modifications, and in 7.5 mm × 10 mm for SIMS analysis. **TPY-Br** was synthesised by a combination of previously described procedures.^{30,35–37} Cyt *c*₅₅₃ 19AA protein was prepared and purified in our laboratory as previously described.³⁸ The ¹H and ¹³C NMR spectra were recorded on a 700 MHz and 176 MHz Agilent DirectDrive2 spectrometer equipped with a room-temperature HCN probe, temperature-controlled at 25 °C. ¹H and ¹³C chemical shifts were calibrated to the residual solvent peak. Coupling constant values (*J*) are given in Hz and chemical shifts (δ) in ppm. The high-resolution electrospray ionisation mass spectroscopy (ESI-HRMS) analyses were performed on a Mariner mass spectrometer (PerSeptive Biosystems).

Synthesis

4'-((4-Bromophenyl)phenyl)-2,2':6',2"-terpyridine tTPY-Br.³⁹ To a solution of 4-bromobenzaldehyde (2.01 g, 10.75 mmol) and 2-acetylpyridine (2.55 mL, 21.47 mmol) in ethanol (50 mL) were added KOH (1.30 g, 23.17 mmol) and NH₄OH (34 mL). The solution was stirred for 4 h until a precipitate appeared. The solution was then filtered, and the solid was washed with ethanol. Recrystallisation of the crude product in a methanol/dichloromethane mixture gave white crystals of the desired compound (1.22 g, 29% yield). ¹H NMR (700 MHz, CDCl₃) δ : 8.73 (d, *J* = 4.5 Hz, 2H), 8.71 (s, 2H), 8.67 (d, *J* = 7.9 Hz, 2H), 7.89 (t, *J* = 7.7 Hz, 2H), 7.78 (d, *J* = 8.4 Hz, 2H), 7.64 (d, *J* = 8.3 Hz, 2H), 7.36 (dd, *J* = 7.0, 5.1 Hz, 2H). ¹³C NMR (176 MHz, CDCl₃) δ 156.2, 156.1, 150.1, 149.2, 137.5, 137.1, 132.3, 129.0,



124.1, 123.6, 121.6, 118.8. HRMS: m/z calc. for $C_{21}H_{15}N_3Br^+$ 388.0449 [$M + H]^+$, found 388.0439.

Diethyl (4-([2,2':6',2"-terpyridin]-4'-yl)phenyl)phosphonate TPY-PO(OEt)₂.⁴⁰ The following compounds were added sequentially to a solution of **tTPY-Br** (500 mg, 1.28 mmol) dissolved in anhydrous toluene (15 mL): diethyl phosphite (0.18 mL, 1.38 mmol), triethylamine (0.2 mL, 1.41 mmol) and Pd(dppf)Cl₂ (37 mg, 0.05 mmol). The reaction mixture was heated at 90 °C under argon for 24 h. The solvent was evaporated to dryness in a vacuum to afford a violet thick oil residue which was crystallised with acetonitrile to afford purple crystals of the desired compound (556 mg, 97% yield). ¹H NMR (700 MHz, CDCl₃) δ : 8.75 (s, 2H), 8.74 (d, J = 4.7 Hz, 2H), 8.68 (d, J = 7.9 Hz, 2H), 7.98 (m, 4H), 7.89 (td, J = 7.7, 1.8 Hz, 2H), 7.37 (ddd, J = 7.4, 4.7, 1.1 Hz, 2H), 4.23–4.09 (m, 4H), 1.36 (t, J = 7.1 Hz, 6H). ¹³C NMR (176 MHz, CDCl₃) δ 156.3, 156.1, 149.3, 142.7, 137.1, 132.6 (d, J = 10 Hz), 129.1 (d, J = 188 Hz), 127.6 (d, J = 14 Hz), 124.1, 121.5, 119.1, 62.4 (d, J = 5 Hz), 16.5 (d, J = 7 Hz). HRMS: m/z calc. for $C_{25}H_{25}N_3O_3P^+$ 446.1634 [$M + H]^+$, found 446.1639.

(4-([2,2':6',2"-terpyridin]-4'-yl)phenyl)phosphonic acid TPY-PO(OH)₂.⁴⁰ Bromotrimethylsilane 97% (0.3 mL, 2.20 mmol) was added to a solution of **TPY-PO(OEt)₂** (270.1 mg, 0.61 mmol) in anhydrous dichloromethane (10 mL) under argon. After stirring the reaction mixture overnight at room temperature, the solvent was evaporated to dryness in a vacuum. Methanol (20 mL) was then added and the solution was stirred at room temperature for 2 h. The reaction mixture was filtered and the solid was washed with methanol and acetone and dried. The obtained solid was dissolved in water (10 mL) by adding NaOH (10% aqueous solution) and precipitated by adding HCl (10% aqueous solution). The solid was filtered off, washed with water, ethanol and acetone and dried to yield a light pink powder of the desired compound (162 mg, 69% yield). ¹H NMR (700 MHz, DMSO) δ : 8.80 (d, J = 4.1 Hz, 2H), 8.78 (s, 2H), 8.73 (d, J = 7.8 Hz, 2H), 8.13–8.03 (m, 2H), 7.90 (dd, J = 12.3, 7.9 Hz, 2H), 7.61–7.56 (m, 2H). ¹³C NMR (176 MHz, DMSO) δ 154.9, 153.8, 149.0, 148.6, 139.3, 138.7 (d, J = 7 Hz), 135.4 (d, J = 181 Hz), 131.6 (d, J = 9 Hz), 126.8 (d, J = 14 Hz), 125.1, 121.6, 118.7. HRMS: m/z calc. for $C_{21}H_{15}N_3O_3P^+$ 388.0851 [$M + H]^+$, found 388.0854.

4'-(4-Pinacolatoboronphenyl)-2,2':6',2"-terpyridine TPY-BO₂-C₆H₁₂.⁴¹ A degassed solution containing **tTPY-Br** (501 mg, 1.29 mmol), bis(pinacolato)diboron (345 mg, 1.36 mmol), KOAc (385 mg, 3.93 mmol) and Pd(dppf)Cl₂ (39 mg, 0.05 mmol) in anhydrous DMSO (6 mL) was stirred at 80 °C for 12 h under argon. Toluene (50 mL) was added and the mixture was washed with water (3 × 100 mL). The organic phase was dried over Na₂SO₄ and the solvent removed by rotary evaporation to yield a light brown solid of the desired compound (529 mg, 94% yield). ¹H NMR (700 MHz, CDCl₃) δ : 8.76 (s, 2H), 8.74 (d, J = 4.4 Hz, 2H), 8.68 (d, J = 7.9 Hz, 2H), 7.95 (d, J = 8.0 Hz, 2H), 7.92 (d, J = 8.0 Hz, 2H), 7.89 (t, J = 7.7 Hz, 2H), 7.36 (dd, J = 6.9, 5.2 Hz, 2H), 1.39 (s, 12H). ¹³C NMR (176 MHz CDCl₃) δ 156.4, 156.1, 150.3, 149.3, 141.1, 137.1, 135.5, 126.7, 124.0, 121.6, 119.1, 84.1, 25.2, 25.1. HRMS: m/z calc. for $C_{27}H_{27}BN_3O_2^+$ 436.2196 [$M + H]^+$, found 436.2198.

Bis(2,2':6',2"-terpyridin-4'-yl)phenyl L φ .⁴² The following compounds were added sequentially to a degassed solution of **TPY-BO₂-C₆H₁₂** (120 mg, 0.28 mmol) and **TPY-Br** (72 mg, 0.23 mmol) in THF (20 mL): a degassed solution of sodium carbonate (102 mg, 0.97 mmol) in water (5 mL) and [Pd (PPh₃)₄] (26 mg, 0.02 mmol). The mixture was refluxed under argon for 48 h. The suspension was then cooled down to room temperature, and then filtered. The white solid was washed with THF and dried to yield the desired compound (124 mg, 99% yield). ¹H NMR (700 MHz, CDCl₃) δ : 8.81 (s, 4H), 8.77 (d, J = 4.2 Hz, 4H), 8.70 (d, J = 8.0 Hz, 4H), 8.07 (s, 4H), 7.90 (t, J = 7.3 Hz, 4H), 7.40–7.36 (m, 4H). ¹³C NMR (176 MHz CDCl₃) 156.4 (TpyC^{6',2'}), 156.2 (TpyC^{2',2'}), 152.1 (ArC^{1,4'}), 149.3 (TpyC^{6,6'}), 139.4 (TpyC^{4'}), 137.1 (TpyC^{4,4'}), 128.1 (ArC^{2,3,5,6'}), 124.0 (TpyC^{5,5'}), 121.6 (TpyC^{3,3'}), 119.0 (TpyC^{3',5'}). HRMS: m/z calc. for $C_{36}H_{25}N_6^+$ 541.2141 [$M + H]^+$, found 541.2153.

ITO cleaning and activation

Commercial ITO surfaces were first cleaned by a detergent/solution (DSC) process; surfaces were scrubbed with a drop of Triton X-100, followed by successive sonication in diluted Triton X-100 solution in water, Mili-Q water and ethanol for 15 min each. Dried surfaces were then activated in an oxygen-fed plasma cleaner (Plasma processing reactor centre Dionex 2000 coupled to a solid-state power generator OEM-12A) operating at 100 W for 15 min, followed by direct immersion in the anchor solution.

Anchoring optimisation

Cleaned and activated ITO surfaces were immersed in 1 mM solution of the different surface anchors overnight (**TPY-PO(OEt)₂** in CHCl₃, **TPY-PO(OH)₂** in DMF : H₂O (1 : 1) and **TPY-PO(OH)₂** in DMSO), then rinsed thoroughly with their respective solvent followed by ethanol and annealed at 140 °C for 30 min to complete the chemical bonding. Surfaces were then sonicated with 5% Et₃N solution in ethanol for 15 min and rinsed generously with ethanol. The anchored ITO surfaces were immersed in 0.1 M solution of FeSO₄·7H₂O in water for 15 min followed by a copious rinsing with water and ethanol. The dried surfaces were then immersed in 1 mM solution of 2,2':6',2"-terpyridine in CHCl₃ for 15 min followed by a generous rinsing with CHCl₃.

Stepwise formation of the modified ITO electrode

Cleaned and activated ITO surfaces were immersed in 1 mM solution of **TPY-PO(OH)₂** in DMSO overnight, then rinsed thoroughly with DMSO followed by ethanol and annealed at 140 °C for 30 min. Surfaces were then sonicated with 5% Et₃N solution in ethanol for 15 min and rinsed generously with ethanol. The anchored ITO surfaces were immersed in 0.1 M solution of the corresponding metal (FeSO₄·7H₂O or Co (NO₃)₂·6H₂O) in water for 15 min followed by a copious rinsing with water and ethanol. The dried surfaces were then immersed in 0.1 mM solution of the ditopic ligand **L φ** in CHCl₃ overnight followed by a generous rinsing with CHCl₃. The metalation step was reproduced as before with the corres-



ponding metal salt. Biofunctionalisation was performed by incubating the modified ITO electrodes with 30 μ M 19AA cyt c_{553} solution (25% glycerol in 5 mM phosphate buffer) for 2 h at room temperature followed by rinsing with 5 mM phosphate buffer (pH 7).

Secondary ion mass spectrometry (SIMS)

SIMS measurements were performed employing a CAMECA SC Ultra instrument under ultra-high vacuum (UHV), usually of 4×10^{-10} mbar. The Cs + primary beam was rastered over $250 \times 250 \mu\text{m}^2$ (the analysis area was limited to $200 \times 200 \mu\text{m}^2$) and positive ion detection mode was used in the experiments and thus all species were measured as CsX + cluster ions. The intensity of the primary beam was 2 pA, and the impact energy was 5 keV. To perform lateral imaging measurements a highly uniform beam was required – the beam on the sample in the SC Ultra tool has a square shape and owing to the “variable rectangular shape concept” forms a homogeneous spot. The primary beam at the working point in the SC Ultra is formed by two stencils – well-shaped apertures. While the first one is used to choose the most intense and homogeneous part of the Gaussian-shaped ion beam, the second one changes the size of the spot.

X-ray photoelectron spectroscopy (XPS)

XPS experiments were carried out using a SPECS Surface Nano Analysis GmbH (Berlin, Germany) instrument equipped with an XR 50 MF X-ray source and a μ -Focus 600 monochromator (600 mm Rowland Circle), using a monochromatised X-ray Al K α emission line, photon energy 1486.6 eV, operating at 100 W, and a Phoibos 150 hemispherical analyser with a 150 mm radius, NAP version, equipped with a 2D-DLD detector. The system base pressure was in the 10^{-10} mbar range. Spectra were fitted using Gaussian–Lorentzian lineshapes using CasaXPS software, version 2.3.18PR1.0 and the quantitative analysis was performed using CasaXPS built-in Scofield relative sensitivity factors.

Electrochemistry

Electrochemical experiments were performed with a Metrohm Autolab B.V. potentiostat/galvanostat in a custom-made Teflon three-electrode cell. Cyclic voltammetry was conducted under an argon atmosphere, using a glassy carbon rod as a counter electrode (CE) and an ITO surface as a working electrode (WE) connected with a conductive adhesive copper tape (6.4 mm width, 1181, 3 M) to provide electrical contact. For non-aqueous experiments, 0.1 M HFPTBA in CH₃CN was used as the electrolyte support with an Ag/AgNO₃ (0.01 M AgNO₃) reference electrode (REF1). The reference electrode was calibrated with ferrocene prior to use with a Fc/Fc⁺ redox potential at 81 mV. For aqueous measurements, 0.1 M phosphate buffer (pH 7) was used as the electrolyte support with an Ag/AgCl (3 M KCl) reference electrode (REF2). The surface coverage Γ (mol cm⁻²) was calculated using the following eqn (1), where I_p is the current peak intensity, n is the number of electrons involved in the process ($n = 1$), F is the Faraday constant, R is

the gas constant, T is the temperature (298 K), ν is the scan rate and A is the surface.

$$I_p = \left(\frac{n^2 \times F^2}{4 \times R \times T} \right) \Gamma \times \nu \times A \quad (1)$$

Electrochemical impedance measurements were conducted in 0.1 M phosphate buffer (pH 7) with 1 mM 1,1'-ferrocenedimethanol in a frequency range from 0.01 Hz to 0.1 MHz. Z-View 2 software was used to fit the impedance data with the equivalent circuit. Photoelectrochemical experiments were performed using a KL 2500 LCD halogen white light source (Schott) with a light intensity of 100 mW cm⁻². The geometric surface area of the analysed samples was calculated as 0.4185 cm². Photochronoamperometric experiments were performed under aerobic conditions at room temperature with a 5 mM phosphate buffer (pH 7) as the electrolyte support. Before each measurement, the open circuit potential (OCP) was recorded under dark conditions until a stable potential was achieved. During photochronoamperometric measurements, samples were illuminated at different potentials (vs. Ag/AgCl) with 30 s ‘light ON/OFF’ periods. The influence of the presence of O₂ was studied firstly in deaerated solution (20 min bubbling with argon), then with a freshly aerated electrolyte solution.

Computational methods

MD setup. Molecular dynamics (MD) simulations have been performed on a cubic $10 \times 10 \times 10 \text{ nm}^3$ box in which 100 TPY molecules have been inserted in a crystalline-like pattern to represent full coverage of the self-assembled monolayer (SAM). One of the TPY molecules has been coordinated with the His₆-tagged 19AA cyt c_{553} . The box has been neutralised with Cl⁻ ions and filled with water molecules. In the starting position the haem was in a perpendicular orientation with respect to the TPY-Fe molecules main axis and 42 Angstroms from the SAM to allow freedom of movement before adsorption on the SAM.

A 15 ns-long NPT molecular dynamics was run with a 2 fs Time step using Desmond software (Desmond Molecular Dynamics System, D. E. Shaw Research, New York, NY, 2020. Maestro-Desmond Interoperability Tools, Schrödinger, New York, NY, 2020) with the OPLS2005 forcefield.⁴³ The Nose–Hoover thermostat was used, keeping the temperature at 300 K, while the pressure was constrained using the Martyna–Tobias–Klein barostat at 1 bar. Long range non-bonding interactions (Coulomb and van der Waals) have been considered within a 1.5 nm cutoff. Water was modelled using the TIP3P model. TPY molecules forming the SAM were restrained with a force constant of 10 Hartree. To speed up the computations, the ITO surface was not considered explicitly, and hydrogen atoms were added to TPY to avoid radicals.

QM/MM setup. The last frame of the MD simulation has been extracted and considered as input for the QM/MM calculations. The TPY-Fe/Haem pair with the shortest distance was considered in this part, with an average distance of 1 nm. Three water molecules completing the metal coordination of



the TPY molecule have been explicitly considered, while the entire protein has been considered in the electrostatic embedding scheme as point charges. Single point calculations with Jaguar 9.5 software⁴⁴ (Jaguar, version 9.5, Schrodinger, Inc., New York, NY, 2017) have been performed using the CAM-B3LYP-D functional and the LACV3P** basis set. All iron metal ions have been considered in the Fe(II) oxidation state, with the low spin state, which led to a +2 formal charge of the entire system. In the case of the Co(II) complexation we used the same snapshot and replaced one of the Fe(II) ions with the Co(II) ion to yield a diradical system.

Results and discussion

Synthesis of the TPY ligands

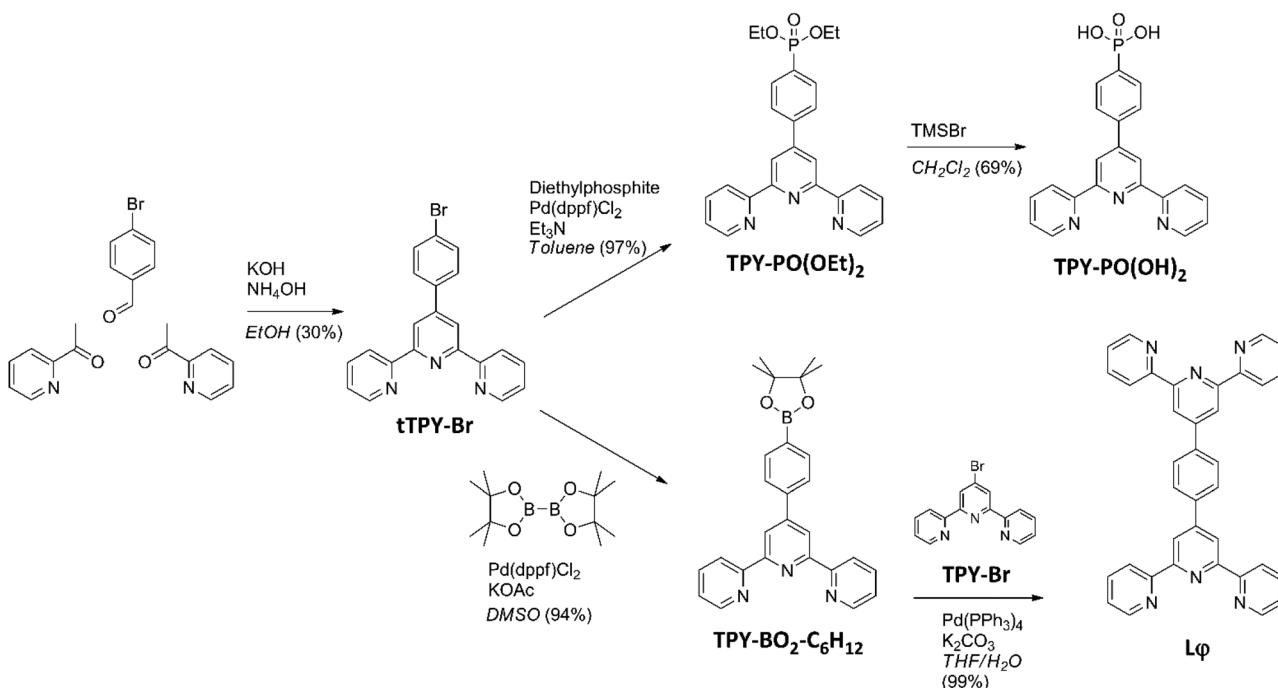
The synthesis pathway used to obtain the anchor **TPY-PO(OH)₂**⁴⁰ and the ditopic ligand **L φ** used in the present study is shown in Scheme 1. The **tTPY-Br** intermediate was formed by a Hantzsch reaction in the presence of potassium hydroxide and ammonium hydroxide in ethanol.³⁹ The phosphonate **TPY-PO(OEt)₂** was obtained by a palladium-catalysed cross-coupling between **tTPY-Br** and diethylphosphite in toluene with triethylamine. As the next step, deprotection was performed using bromotrimethylsilane in dichloromethane. The boronate ester **TPY-BO₂-C₆H₁₂** was obtained from a palladium-catalysed cross-coupling between **tTPY-Br** and bis(pinacolato)diboron in dimethylsulfoxide in the presence of potassium acetate.⁴¹ Finally, the ditopic ligand **L φ** (Fig. S1–S4†) was synthesised by a Suzuki–Miyaura coupling between **TPY-BO₂-C₆H₁₂** and

TPY-Br in a water-tetrahydrofuran mixture in the presence of potassium carbonate.⁴²

Optimisation of the ITO surface anchoring process

Although the phosphonic-type attachment group showed efficient properties concerning stability and reactivity towards different types of electrode surfaces,^{45–47} the compound **TPY-PO(OH)₂** has been used in this study for the first time for direct anchoring onto the ITO surface. In 2010, Spampinato *et al.* reported a methodology to graft this anchor on an SiO_2 surface using a priming step based on a zirconium phosphate monolayer. The anchor can then be attached either by the TPY moiety or by the phosphonic acid.⁴⁰ In 2007, Wolpher *et al.*, described properties of ruthenium(II) complexes grafted on nanostructured TiO_2 .⁴⁸ However, the authors of the latter study directly attached the entire complex onto the surface in contrast to the step-by-step, bottom-up approach of the present study.

In order to find the best conditions for ITO grafting, both **TPY-PO(OEt)₂** and **TPY-PO(OH)₂** compounds were investigated since both functions were reported for efficient attachment to various types of electrode surfaces *via* a phosphonate group.^{49–53} Commercially available ITO surfaces were first subjected to a detergent/solution cleaning (DSC) followed by an activation with oxygen plasma treatment to improve electroactivity and surface work function.⁵⁴ The plasma-activated surfaces were functionalised using different conditions depending of the compound solubility: **TPY-PO(OEt)₂** in CHCl_3 , **TPY-PO(OH)₂** in $\text{DMF} : \text{H}_2\text{O}$ (1 : 1)⁴⁰ or **TPY-PO(OH)₂** in DMSO , followed by an annealing treatment to increase chemical bond stability.^{55,56}



Scheme 1 Synthesis pathway for the anchor **TPY-PO(OH)₂** and the ditopic ligand **L φ** .



Incorporation of the redox metal centre was performed in an aqueous solution of $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$ and the coordination sphere was finally completed by immersing the surfaces in a solution of 2,2':6',2''-terpyridine.

To assess the efficiency of surface coverage using the above-mentioned step-by-step approach, the freshly modified electrodes were analysed by secondary ion mass spectrometry (SIMS) to simultaneously map the presence of phosphorus and iron atoms. The SIMS analyses confirmed that efficient TPY anchoring and complex formation was accomplished on the ITO surface especially when DMSO was used as the solvent for the TPY-PO(OH)_2 modifier (Fig. 1C). The mapping result obtained for TPY-PO(OEt)_2 shows a strong heterogeneity between phosphorus and iron distribution with the non-functionalised area corresponding to the black spots in Fig. 1A. The higher concentration of phosphorus compared to iron could be rationalised by a partial protonation of the terpyridine or a misalignment of the terpyridine motif due to the acidic tendency of chloroform⁵⁷ preventing the chelation of the iron centre. In contrast, both samples prepared from TPY-PO(OH)_2 display better surface coverage (Fig. 1B and C) characterised by a higher concentration of phosphorus atoms. A particularly higher density and clear coverage homogeneity of phosphorus and iron atoms are demonstrated for the sample produced from DMSO solution (when compared to other solvents) displaying the presence of the same subtle shadow pattern in both phosphorus and iron maps (Fig. 1C). The observed mismatch between phosphorus and iron atoms in the case of the DMF : H_2O solution could be explained by the poor solubility of the anchor, and the presence of residual compounds, which could hinder the access of iron.

A similar sample of TPY-PO(OH)_2 in DMSO with cobalt was also analysed (Fig. S6†), confirming the good surface coverage and efficient formation of cobalt complexes. Consequently, all following experiments on the functionalisation of the ITO electrodes were performed using TPY-PO(OH)_2 solution in DMSO.

Construction and characterisation of cobalt and iron-based ITO electrodes

The methodology employed for the construction of *c*₅₅₃ bio-hybrid electrodes is depicted in Fig. 2. It was important to rationally design the final TPY molecular structure so as to achieve fine-tuning of the ET process within the SAM through incorporation of the metallic redox centres into the wire. To this end, two different metals were chosen for insertion into the structure of the TPY molecules, iron and cobalt, for their ability to easily form TPY complexes under mild conditions, which is optimal for the development of a viable, step-by-step, bottom-up strategy. Typically, cleaned and activated ITO surfaces were anchored with TPY-PO(OH)_2 in DMSO, as previously described. Metalation was performed with either $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$ or $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ aqueous solution followed by the complexation with the ditopic ligand **Lφ** in CHCl_3 . Homo-dinuclear TPY wires were formed by a second metalation step. Finally, the *c*₅₅₃ protein molecules (Fig. S6†) were immobilised through binding of the C-terminal His₆-tag moiety (genetically incorporated into the *c* structure, see the ESI†) to the metal centre in order to form a thin layer of this electroactive protein.

X-ray photoelectron spectroscopy (XPS) analysis was performed on the bare ITO surface and on the full organometallic wires (ITO-TPY-Co-Lφ-Co and ITO-TPY-Fe-Lφ-Fe) in order to analyse the elemental composition of the surfaces (Fig. S7 and

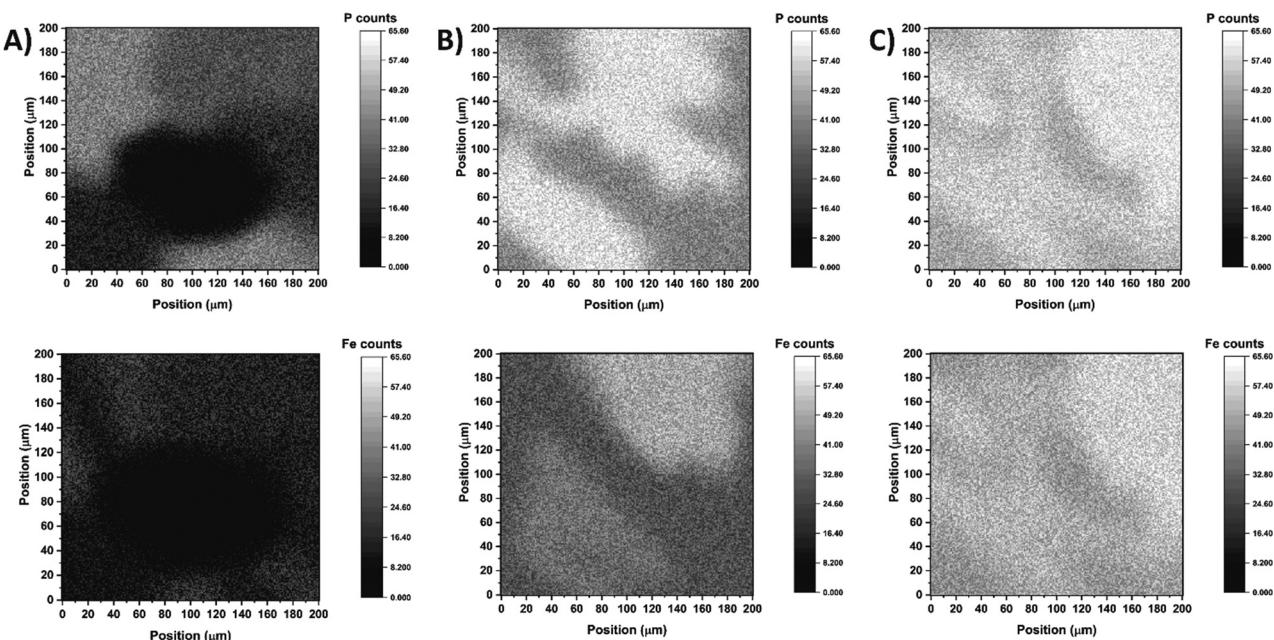


Fig. 1 Secondary ion mass spectrometry analyses of ITO-TPY-Fe-TPY electrodes with phosphorus (top) and iron (bottom) atoms. (A) Surface obtained with TPY-PO(OEt)_2 in CHCl_3 . (B) Surface obtained with TPY-PO(OH)_2 in $\text{DMF} : \text{H}_2\text{O}$ (1 : 1). (C) Surface obtained with TPY-PO(OH)_2 in DMSO.



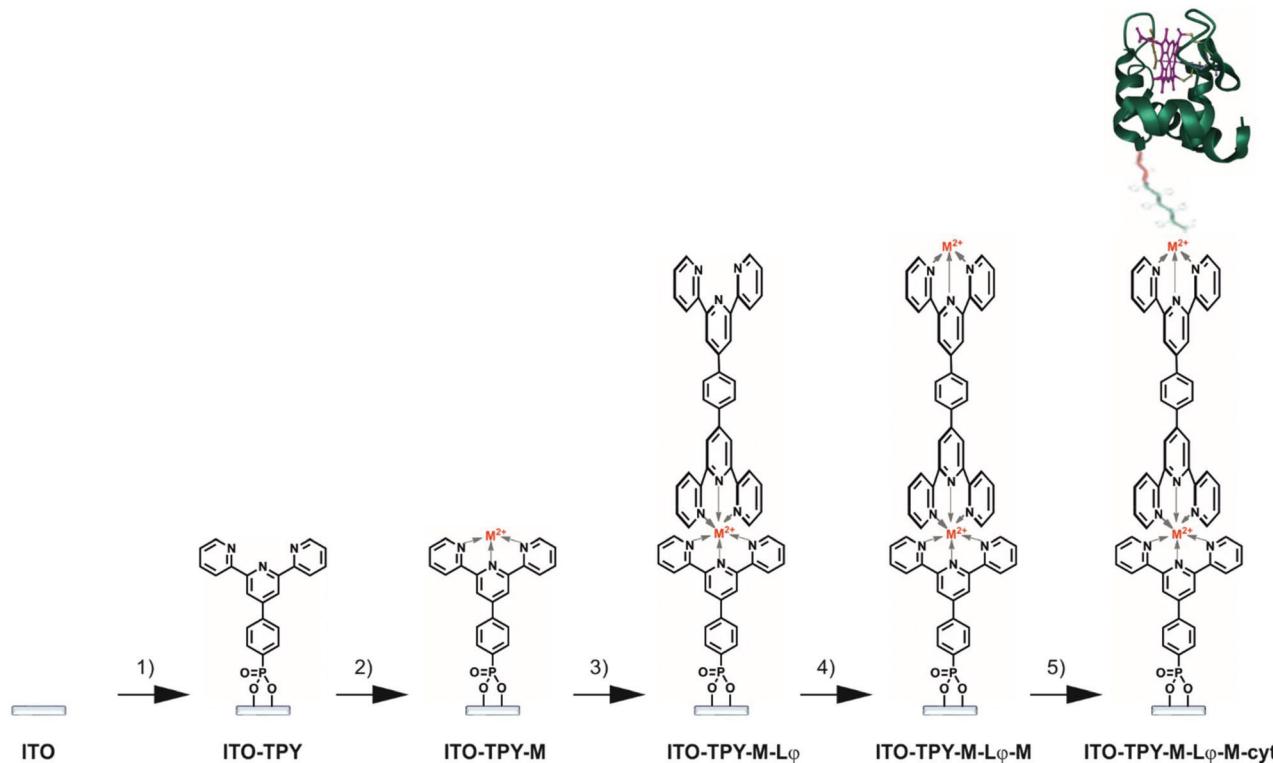


Fig. 2 Stepwise preparation of functionalised ITO electrodes. (1) 1 mM TPY-PO(OH)₂ in DMSO. (2) 0.1 M aqueous solution of FeSO₄·7H₂O or Co(NO₃)₂·6H₂O. (3) 0.1 mM Lφ in CHCl₃. (4) 0.1 M aqueous solution of FeSO₄·7H₂O or Co(NO₃)₂·6H₂O. (5) 30 μM 19AA cyt c₅₅₃ in phosphate buffer.

S8†). Due to the extremely low concentration of Co-TPY and Fe-TPY on the ITO surface, the registered spectra for those samples were dominated by signals of ITO components. Analysis of the Fe 2p region for the pure ITO and ITO-TPY-Fe-Lφ-Fe samples revealed overlapping of potential Fe 2p signals with strong In 2p_{1/2} and Sn 3p_{3/2} signals, making the detection of Fe impossible (Fig. S7†). The small Fe 2p signals are completely masked by the In 3p_{1/2} and Sn 3p_{3/2} signals.

The analysis of the Co 2p region (Fig. S8A†) in the ITO-TPY-Co-Lφ-Co sample revealed the Co 2p signals typical of Co(II) components, namely 2p_{3/2} and 2p_{1/2} doublets at 781.5 and 796.6 eV, respectively (*versus* 778.3 for Co 2p_{3/2} for metallic Co⁵⁸), and two doublets of satellite peaks (present only for the oxidised Co) at 787.0 and 802.1 eV (first doublet) and 792.1 and 806.6 eV (second doublet), together with the Co LMM signal at 776.0 eV. Those signals were not present in the case of pure ITO (see Fig. S8A†). XP spectra registered for N 1s and P 2p regions (Fig. S8B†) allowed for a rough approximation of sample elemental composition. As both samples are dominated by In and Sn (components of ITO), O (component of ITO and part of the adventitious organic contamination), and C (mostly adventitious carbon), the general atomic composition did not represent the Co-TPY composition. Thus, we focused on Co, N and P ratios, as those elements are not common contaminants and were not present in significant amounts in the control ITO sample (see Fig. S8†). Using integrated Co 2p, N 1s and P 2p signals (Fig. S8†) and Casa build-in Scofield Relative

Sensitivity Factors, equal to 1.8 for N 1s, 1.192 for P 2p and 19.16 for Co 2p, the Co:N:P atomic ratio was determined as close to 1.0:2.3:1.0, where the small amount of N and P originating from the reference ITO sample was subtracted from the N and P amounts determined from the N 1s and P 2p signals in the Co-TPY-ITO sample. The XPS quantitative analysis proved to be challenging due to the fact that a monolayer of TPY ligand wires yielded extremely low signals, and thus, determination of the peak area for such low intensity peaks may result in errors as high as ±25%.

The step-by-step construction of TPY-functionalised electrodes was followed by cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) analyses which were used to explore the evolution of interfacial electronic processes (Fig. 3). Experiments were performed in the presence of a redox probe 1,1'-ferrocenedimethanol in phosphate buffer (pH 7) in a frequency range from 0.01 Hz to 0.1 MHz. The corresponding Nyquist plot and CV scans obtained for Fe-TPY-based electrodes are presented in Fig. 3 (see Fig. S9† for Co-TPY ligand-functionalised electrodes). The charge transfer resistance R_2 and the double layer capacitance C_{dl} of the bare and modified ITO electrodes were calculated by fitting the impedance data with the equivalent circuit model (Fig. 3C) where R_1 stands for the electrolyte resistance, CPE1 is the constant phase element corresponding to double layer capacitance⁵⁹ existing between the solid/liquid phases, R_2 corresponds to the charge transfer resistance between the electrolyte and the

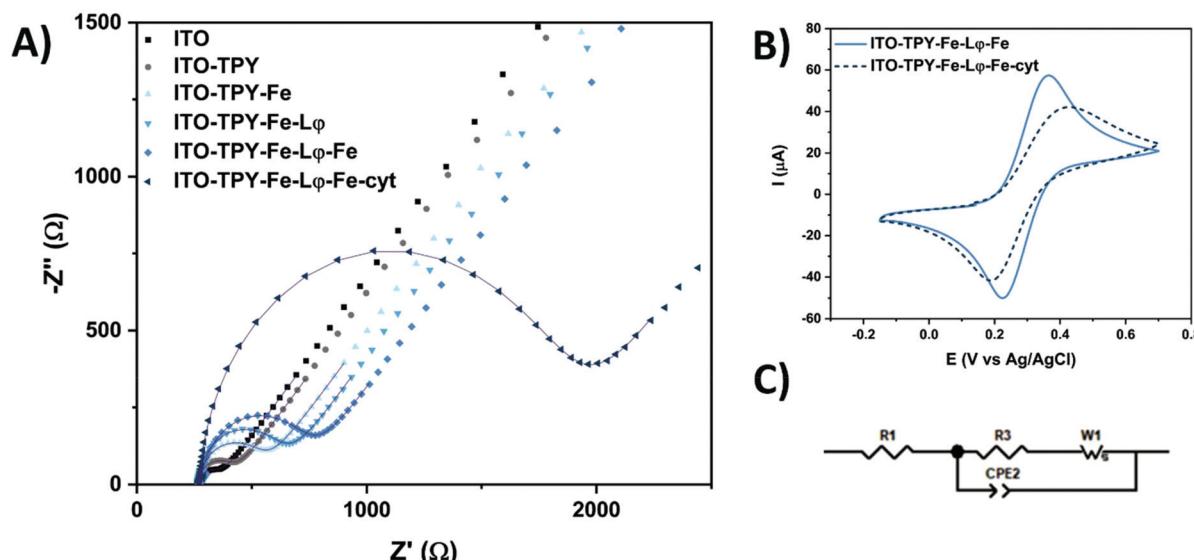


Fig. 3 (A) Nyquist plot with fitted curves of electrochemical impedance spectra obtained for Fe-based ITO electrodes at different steps of functionalisation. (B) Cyclic voltammetry of ITO-TPY-Fe-L φ -Fe and ITO-TPY-Fe-L φ -Fe-cyt at 100 mV s $^{-1}$. (C) Equivalent circuit model. All measurements were performed with 1 mM 1,1'-ferrocenedimethanol in 0.1 M phosphate buffer (pH 7) in the dark.

studied ITO electrode, and W_1 represents Warburg impedance caused by the diffusion process. The obtained resistances (R_1/R_2) and capacitance C_{dl} (from CPE1 parameter Q and ϕ^{59}) values are summarised in Table 1. The evolution of the resistance values shown in Fig. 3A and in Table 1 is consistent with the step-by-step construction of the organometallic TPY ligand molecular wires (terminated with the anchored His₆-tagged cyt c_{553}). The bare ITO electrode possesses a small charge transfer resistance of 178 $\Omega \text{ cm}^{-2}$ in accordance with the good conductivity of this material. After the attachment of TPY-phosphonic acid, the resistance rises to 323 $\Omega \text{ cm}^{-2}$ and continues to increase after each step of functionalisation, reaching the final value of 1.09 $\text{k}\Omega \text{ cm}^{-2}$ and 1.02 $\text{k}\Omega \text{ cm}^{-2}$ respectively for Fe- and Co-containing TPY wires. Even though the construction of the organometallic interface hinders somewhat the charge transfer process, it is interesting to notice that the whole conductivity, which is inversely proportional to the resistivity, is

still promising prior to the cyt c_{553} attachment. After the immobilisation of this protein, a significant increase of resistance is observed with the values of 3.73 $\text{k}\Omega \text{ cm}^{-2}$ for iron and 2.39 $\text{k}\Omega \text{ cm}^{-2}$ for cobalt nanoassemblies, which is correlated with the insulating properties of the dense protein backbone.^{60–62} The hindrance of ET due to the presence of cyt c_{553} is also visible from the CV analysis shown in Fig. 3B. For the Fe-based TPY wires, a well-defined quasi-reversible redox process is observed. After the biofunctionalisation step, a clear attenuation of the electrochemical processes is recorded with an increase of the peak-to-peak potential separation of 100 mV and a decrease of current intensity.

Interestingly, the double layer capacity of the interface does not significantly change upon the stepwise electrode functionalisation suggesting that the nanostructures with inbound cyt c_{553} molecules do not collapse at the electrode surface. Moreover, the ϕ parameter of the CPE element is close to

Table 1 The curve fitting results based on the $R_1(R_2W_1)\text{CPE1}$ model under different illumination conditions in 1 mM 1,1'-ferrocenedimethanol + 0.1 M phosphate buffer (pH 7)

Samples	Dark			Light		
	$R_1 (\Omega \text{ cm}^{-2})$	$R_2 (\text{k}\Omega \text{ cm}^{-2})$	$C_{dl} (\mu\text{F cm}^{-2})$	$R_1 (\Omega \text{ cm}^{-2})$	$R_2 (\text{k}\Omega \text{ cm}^{-2})$	$C_{dl} (\mu\text{F cm}^{-2})$
ITO	672	0.18	12.5	663	0.25	13.2
ITO-TPY	632	0.32	13.3	618	0.42	13.8
ITO-TPY-Co	627	0.61	14.0	642	0.71	14.2
ITO-TPY-Fe	639	0.65	13.8	639	0.77	14.0
ITO-TPY-Co-L φ	628	0.84	14.9	626	0.90	14.8
ITO-TPY-Fe-L φ	628	0.87	14.1	611	0.89	14.3
ITO-TPY-Co-L φ -Co	644	1.02	14.0	638	1.02	14.1
ITO-TPY-Fe-L φ -Fe	664	1.09	14.1	663	1.10	14.4
ITO-TPY-Co-L φ -Co-cyt	632	2.39	14.7	628	2.26	14.7
ITO-TPY-Fe-L φ -Fe-cyt	790	3.73	15.3	774	3.26	15.4

unity ($0.92 < \phi < 0.96$) suggesting the formation of a highly ordered architecture at the molecular level.⁶³ The homogeneous electrode coverage with the bio-interface was confirmed by the calculation of the apparent fractional electrode coverage parameter ϕ ,⁶⁴ with the respective values of 0.93 for Co-TPY wires and 0.95 for the Fe-TPY counterparts.

The interface resistivity was also measured under light conditions for all the different configurations (Table 1). Similar to the bare ITO, all the electrodes devoid of cyt *c*₅₅₃ possess a higher or similar resistance upon exposure to light compared to their respective resistance under dark conditions. In contrast, for both cyt-based electrodes noticeably lower resistivity values were observed, especially for the iron-based TPY interface, with a decrease from $3.73 \text{ k}\Omega \text{ cm}^{-2}$ to $3.26 \text{ k}\Omega \text{ cm}^{-2}$ (see Table 1). The observed lower resistivity in the presence of cyt *c*₅₅₃ highlights the beneficial combination of TPY SAM and the redox protein under light irradiation, which is promising for further development of high-performance bio-photoelectrodes, such as those based on photosynthetic macromolecular machines.^{18,65}

Additional electrochemical analyses were performed in acetonitrile with 0.1 M hexafluorophosphate tetrabutylammonium (HFPTBA) for Fe- and Co-TPY organometallic wires in order to study their redox behaviour upon immobilisation on ITO (Fig. 4 and Table 2). In solution, these TPY complexes possess a M^{III}/M^{II} redox signature at $E_{1/2} = 0.76 \text{ V}$ and $E_{1/2} = -0.06 \text{ V}$,⁶⁶ respectively for iron and cobalt with a peak-to-peak potential separation of 60–70 mV, typical of a reversible system ($n = 1$). Upon their assembly on the ITO surface, the Fe-TPY complexes show a redox signature at $E_{1/2} = 0.81 \text{ V}$ (Fig. 4A) and the Co counterparts at $E_{1/2} = -0.1 \text{ V}$ (Fig. S10†). Their peak-to-peak potential separation values are very small with respective values of 5 mV and 12 mV for Fe and Co complexes, which indicates a surface process with fast ET kinetics and also points out a facile charge transport between the ITO electrode and the TPY SAM.³¹ The high reversibility of the ET process demonstrates the formation of the highly conductive

Table 2 Redox potentials of iron and cobalt terpyridine complexes in solution and assembled on ITO electrodes and their respective calculated surface coverage, in deoxygenated CH₃CN + 0.1 M HFPTBA at 100 mV s⁻¹

Samples	$E_{1/2}$ (V)	ΔE_p (mV)	Γ (mol cm ⁻²)	k_{ET} (s ⁻¹)
tTPY-Co-tTPY ^a	-0.06 ^b	70 ^b	—	—
ITO-TPY-Co-L ϕ	-0.10	12	1.27×10^{-11}	29
tTPY-Fe-tTPY ^a	0.76 ^b	60 ^b	—	—
ITO-TPY-Fe-L ϕ	0.81	5	1.65×10^{-11}	29

^a In solution (CH₃CN + 0.1 TBAP). ^b Data from ref. 57.

TPY SAM on ITO, which is further confirmed by the observed linear dependency of current intensity with the scan rate^{67,68} (Fig. 4B and Fig. S10B†). From these analyses, the surface coverage of both Fe-and Co-TPY SAMs was determined as $1.65 \times 10^{-11} \text{ mol cm}^{-2}$ and $1.27 \times 10^{-11} \text{ mol cm}^{-2}$, respectively, corresponding to the values reported for similar architectures on ITO.^{69,70} Using Laviron's equation,⁷¹ the interfacial electron transfer kinetic constants k_{ET} of both complexes were determined with a similar value of 29 s^{-1} . This electron rate is lower compared to the one published on gold⁷² or silicon⁷³ probably due to the loss of conjugation induced by the tetrahedral phosphorus atom in the TPY anchor. Nevertheless, the kinetic constant values determined for both systems are still promising for the attachment of cytochrome *c* which was shown to yield smaller electron transfer rates when adsorbed⁷⁴ ($k_{ET} = 18 \text{ s}^{-1}$) or wired⁷⁵ ($k_{ET} = 5.9 \text{ s}^{-1}$) onto ITO.

It was important to relate all the above data to the electrochemical properties of the immobilised TPY SAM under biocompatible conditions, as ultimately the developed TPY interface is destined for the domain-specific capturing of (photo) electroactive proteins including photosystems and other redox active enzymes of biotechnological and fundamental applications. To this end, a preliminary electrochemical analysis of the His₆-tagged cyt *c*₅₅₃ in solution was performed using an aqueous buffer (phosphate buffer, pH 7) as an electrolyte and

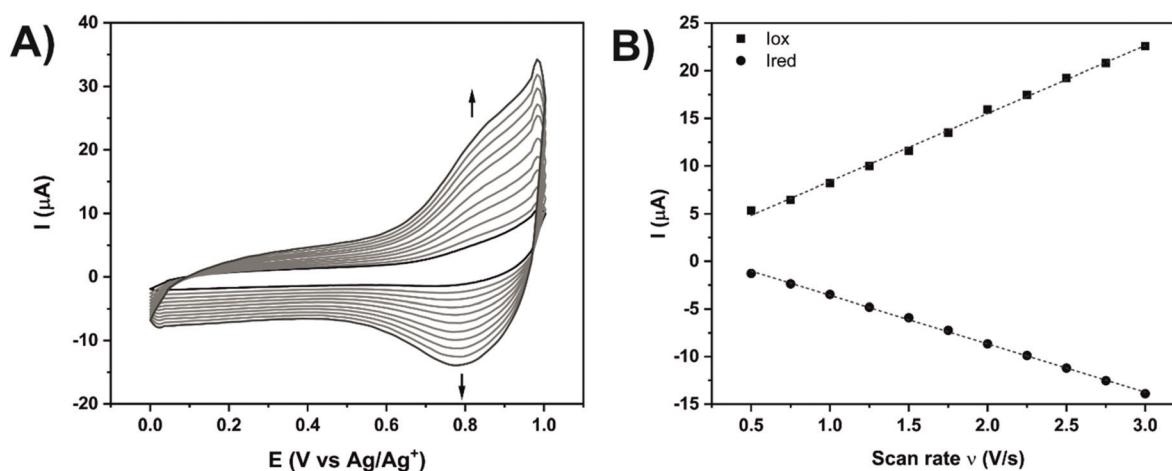


Fig. 4 (A) Cyclic voltammetry of ITO-TPY-Fe-L ϕ at different scan rates from 500 mV s^{-1} to 3 V s^{-1} in CH₃CN + 0.1 M HFPTBA. (B) Linear relation between the current peak and scan rate obtained for ITO-TPY-Fe-L ϕ .



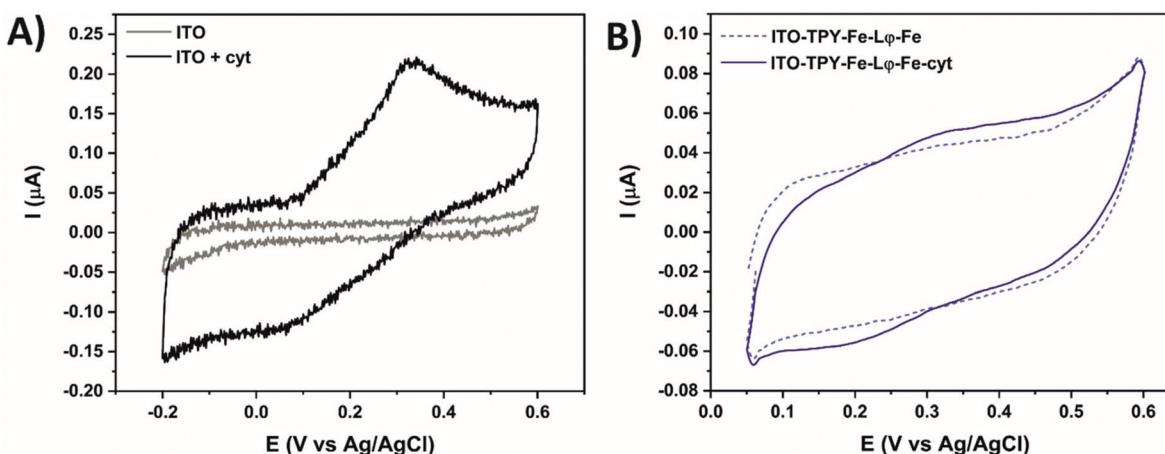


Fig. 5 Cyclic voltammetry at 5 mV s⁻¹ in 0.1 M phosphate buffer (pH 7) of the ITO electrode without and in the presence of 7 μM cyt c₅₅₃ (A). Fe-based electrodes with and without immobilised cyt (B).

a bare ITO as a WE (see Fig. 5A). As expected, in the potential range from -0.2 V to 0.6 V the bare ITO is redox inactive. After the addition of cyt c₅₅₃, a well-defined redox system appears with an oxidative peak at 0.33 V and a reductive peak at 0.06 V. This redox signature, attributed to the Fe^{III}/Fe^{II} couple in the haem group, is positively shifted compared to previously reported data on ITO found between -0.1 V and 0.1 V for horse heart and yeast cyt c.^{74,76,77} Although different, the redox peak detection confirms the ability of the modified cyt c₅₅₃ to conduct ET with pristine ITO. Finally, both metal-based architectures with immobilised cyt were analysed under the same conditions with regards to the 5 mV s⁻¹ scan rate.

Fig. 5B shows the CV of the Fe-based TPY-modified electrodes in the absence or presence of the anchored cyt c₅₅₃. A clear signature of the immobilised redox active cyt protein is visible, with an oxidative peak at 0.33 V and a reductive peak at 0.2 V. The peak-to-peak potential separation of 130 mV is smaller compared to the one found for cyt c₅₅₃ in solution (270 mV), which proves the immobilisation of the protein with an improvement of surface processes. Moreover, the smaller peak-to-peak potential separation proves the structural integrity of the immobilised cytochrome, whereas the formation of aggregates or unfolded cytochrome was shown to result in fully irreversible CV signatures.^{78,79} Concerning the Co-based nanoarchitecture, the electrochemical analyses revealed quenching of the ET process as confirmed by the absence of the cyt c₅₅₃ redox signature (Fig. S11†).

The difference in the ET properties depending on the metal-centre used could be explained by a variable contribution of the ET depending on the metal centre used in the TPY SAM. In the case of the iron, this metal centre facilitates ET between the haem group of cyt and the ITO surface, whereas the incorporation of a cobalt redox centre into the TPY SAM inhibits this process. These electrochemical data clearly point to the importance of the SAM molecular structure for efficient ET to occur in the biohybrid electronic devices, whereby the presence of the specific metallic centre in the con-

ductive SAM can greatly affect the kinetics and directionality of ET.⁸⁰

Photocurrent measurements

In order to verify the suitability of the organometallic TPY-based SAM developed in this study for future applications, such as those involving the oriented light-harvester/charge separator, photosystem I,¹⁸ photochronoamperometry measurements were conducted for all the electrode configurations. All the analyses were conducted in an oxygenated aqueous buffer without any external electron mediator. The obtained photocurrent density values (J) are presented in Fig. 6.

Overall, all the TPY-functionalised electrodes are characterised by a preferential generation of the cathodic photocurrent. This feature is particularly prominent in the case of Fe-TPY SAM. Indeed at +100 mV, the highest anodic current was recorded for Fe-based TPY wires with a value of 17.2 nA cm⁻²,

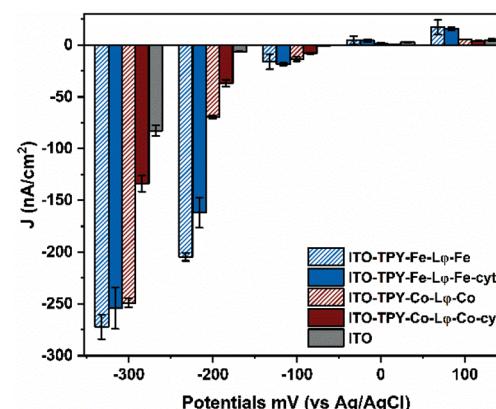


Fig. 6 Photocurrent density generation of ITO electrodes at different applied potentials vs. Ag/AgCl. The current density values correspond to the average values from two independent measurements with error bars representing standard deviation (n = 2).



slightly above that of the corresponding electrode with captured cyt *c*₅₅₃ (15.6 nA cm⁻²). At 0 mV, all electrodes behave similar to the bare ITO electrode with values oscillating between 0.7 to 4.4 nA cm⁻². In the negative potential range, the presence of the Fe-TPY SAM yields a significant improvement of cathodic photocurrent generation, exhibiting over 20-fold higher photocurrent output compared to a bare ITO control (\approx 17 nA cm⁻² *vs.* 0.7 nA cm⁻² at -100 mV). Interestingly, at -200 mV a photocurrent of \sim 200 nA cm⁻² was recorded for Fe-TPY-based electrodes, with a respective 30-fold and 24-fold higher production without and with cyt *c*₅₅₃, respectively.

In comparison, the photocurrent production of Co-based wires is less pronounced at this potential, with an obvious difference between the electrodes without and with cyt *c*₅₅₃ (69.8 nA cm⁻² *vs.* 36.8 nA cm⁻²). This trend is even more visible at -300 mV. While both Fe- and Co-based TPY-functionalised electrodes generate a similar photocurrent of \sim 260 nA cm⁻², the addition of cyt *c*₅₅₃ onto the Co-based electrode results in a 2-fold decrease of the cathodic photocurrent. These observations are in accordance with the previous electrochemical analyses and confirm the role of the metal in the electron transfer process through the TPY wire in these nano-assemblies. At positive and negative potentials (Fig. S11† and Fig. 6), the presence of cobalt quenches the ET process between the cyt and the ITO electrode, whereas in the case of iron the metal centre facilitates the ET process (Fig. 5B and 6).

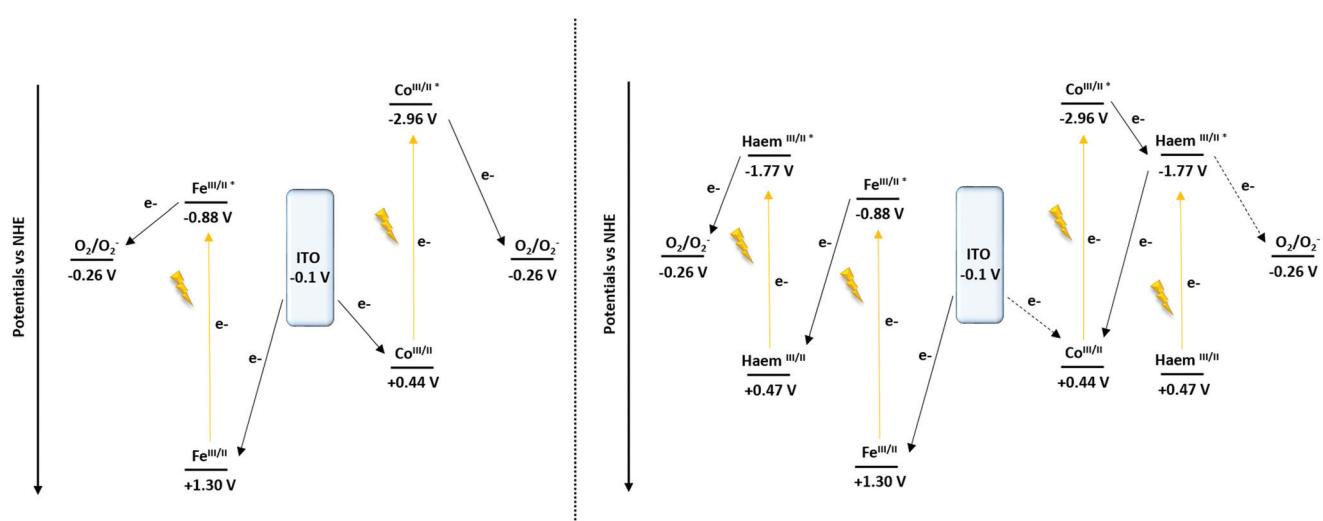
The observation of the ET process between ITO and cyt *c* that is promoted by Fe(II) embedded in the structure of TPY ligand molecular wire is closely related to the electron transfer step exerted by the easily oxidizable non-haem Fe present on the electron acceptor side of Type II photosynthetic reaction centres (photosystem II and the bacterial reaction centre⁸¹). In these macromolecular structures the non-haem Fe mediates ET occurring on a μ s timescale^{82,83} between the two quinones

(Q_A and Q_B) that are symmetrically located on either side of this metal centre. The removal of Fe or its replacement with other metals (*e.g.* Zn, Co) has been shown to alter the electron transfer rates from the intermediate acceptor to the primary acceptor Q_A,^{84,85} highlighting the importance of the non-haem iron for the native ET processes on the electron acceptor side of the photosynthetic reaction centres.

Previous electrochemical studies on other types of SAMs supported by DFT calculations explained that the electron flow can be tuned by the presence of different metal redox centres.^{80,86} In this case, the observed decrease in the photocurrent generation in the presence of cyt *c*₅₅₃ can be explained by the interplay between the cathodic and anodic currents occurring simultaneously. In fact, for all the configurations, ET occurs from the electrode to the organometallic SAM (cathodic current). With the addition of cytochrome, the cathodic electron flow is preserved for Fe-based TPY SAM, whereas for a Co-containing nanoconstruct a competitive reverse electron flow (from cyt *c*₅₅₃ to ITO) occurs (anodic current). Nevertheless, the effectiveness of covalently attached TPY wires for cathodic photocurrent generation is well demonstrated in the present work exceeding the photocurrent output obtained for other cyt-modified electrodes,⁸⁰ even after plasmonic enhancement (92.3 nA cm⁻² *at* -200 mV)⁶⁵ or for PSI-based ITO electrodes in the presence of an external mediator (160 nA cm⁻² *at* -300 mV).¹⁶

To support the hypothesis concerning the influence of the cyt *c*₅₅₃ on the cathodic photocurrent generation, the energy diagrams of the Co- and Fe-based configurations were determined⁸⁷ (Scheme 2).

From the converted redox potentials of the Co- and Fe-TPY complexes immobilised on ITO *vs.* the SCE reference electrode (by adding 298 mV (ref. 88) to the values *vs.* Ag/Ag⁺), the energy of the highest occupied molecular orbital (HOMO) can be calculated. The respective values of -4.94 eV ($+0.44$ V *vs.* NHE) and -5.80 eV ($+1.30$ V *vs.* NHE) related to the vacuum



Scheme 2 Proposed mechanism of the cathodic photocurrent recorded at -0.1 V *vs.* NHE (-0.3 V *vs.* Ag/AgCl) of the ITO-TPY devices without (left) and with the presence of immobilised cyt *c*₅₅₃ (right).



energy level were determined using eqn (2) with an energy of -4.74 eV for SCE with respect to the zero-vacuum level. Considering the 0–0 transition energy ΔE^{00} of the metal-terpyridine chromophores ($\lambda_{\text{Co-TPY}}^{00} = 365$ nm (ref. 87) and $\lambda_{\text{Fe-TPY}}^{00} = 568$ nm (ref. 66)), the energy of the lowest occupied molecular level (LUMO) can be estimated at -1.54 eV (-2.96 V vs. NHE) and -3.62 eV (-0.88 V vs. NHE), respectively, related to the vacuum energy level using the eqn (2) and (3).

$$E_{\text{HOMO}} \text{ (eV)} = -4.74 - E_{1/2} \quad (2)$$

$$\Delta E^{00} = E_{\text{LUMO}} - E_{\text{HOMO}} = 1240/\lambda^{00} \quad (3)$$

Similarly for cyt c_{553} , after the conversion of the redox potential vs. SCE (by removing 39 mV to the value vs. Ag/AgCl) and with the 0–0 transition energy ($\lambda^{00} = 553$ nm, Fig. S6†), the energy levels of the HOMO and the LUMO were found at -4.97 eV ($+0.47$ V vs. NHE) and -2.72 eV (-1.77 V vs. NHE), respectively.

Concerning the nano-architectures devoid of cyt c_{553} (Scheme 2, left), under illumination ($370 \text{ nm} < \lambda < 790 \text{ nm}$) both metal-TPY complexes absorb light and are promoted to their respective excited states. In the next step, the electron transfer process occurs, ultimately yielding the reduction of oxygen molecules present in the electrolyte solution, as oxygen is a well-known electron acceptor.^{89,90} Finally, the initial redox state of the devices is regenerated by electron transfer from ITO, which results in a photo-induced cathodic current generation enhanced by the presence of O_2 (Fig. S12†).

In the case of the Fe-based biohybrid devices (Scheme 2, right), the photo-induced electron mechanism depicts a “Z-scheme” similarly to the natural photosynthesis. Upon illumination the haem and Fe-TPY complex absorb light and are promoted to the excited states. While the excited state of the haem is able to reduce the oxygen (Fig. S12†), a favourable electron transfer occurs from the excited state of the Fe-TPY complex to regenerate the ground state of the haem. Finally, the ground state of the device is restored by the electron injection from the ITO substrate. On the other hand, for the Co-based nanosystem the presence of the haem induces a competitive ET process. After the excitation of the cobalt TPY complex, an electron transfer occurs to the excited state of the haem group from which point two mechanisms are possible: (1) either the electron is transferred to oxygen (Fig. S12†) or (2) a back-electron transfer (BET) reaction takes place to regenerate the initial redox state of the cobalt; thus, reducing the overall cathodic photocurrent output.

Computational analysis of the electron transfer mechanism

The QM/MM calculations reveal that for the TPY-Fe/haem pair frontier orbitals are localised over two different fragments, with the HOMO localised over the TPY molecule and the LUMO localised on the haem (Fig. 7). The energy of the HOMO was found at -4.70 eV and for the LUMO at -4.28 eV, leading to an energy gap of 0.42 eV. For the TPY-Co/haem interface, the situation is the opposite. The presence of cobalt ion leads to the presence of a diradical state of the TPY molecule, resulting in a different localisation of the frontier molecular orbitals, with the two semi-

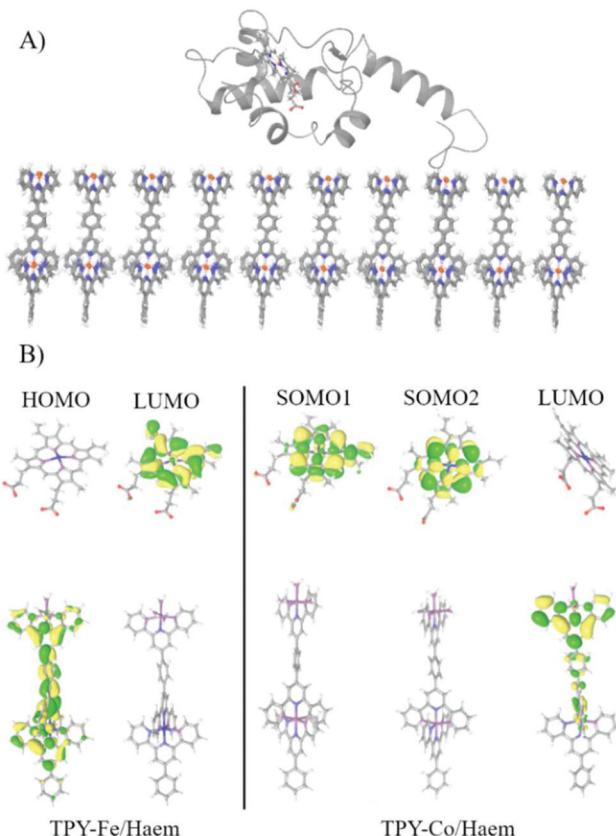


Fig. 7 (A) The last snapshot of MD simulation showing the equilibrated interface. (B) Frontier molecular orbitals for TPY-Fe/haem and TPY-Co/haem interfaces.

occupied SOMOs localised over the haem group and the LUMO localised over the TPY molecule. The energy for the SOMO are -4.55 and -4.39 eV for spin up and down, respectively, while the LUMO is found at -4.00 eV, leading to an energy gap of 0.39 eV. The computational results may be used to explain the difference in photogenerated current observed experimentally. When applying a cathodic bias (from ITO to cyt c_{553}) the current is enhanced for the TPY-Fe/haem interface, as the internal electron flow and the external bias are parallel. For the TPY-Co/haem complex, the situation is different as due to the triplet states recombination⁹¹ a diminished ET is observed. This results in a lower photogenerated current for the TPY-Co/haem interface with respect to the TPY-Fe/haem counterpart. In addition, the energy gap found for both interfaces is relatively low, opening the possibility of an external bias, even small, to overcome the internal flux of electrons, as seen for the anodic bias.

Several possible electronic-coupling pathways can be predicted in the heterogeneous cyt/TPY biomolecular system of this study (Fig. 8). The shortest distance found between haem and the TPY molecule (0.99 nm) is without the mediation of any amino acid present within the structure of cyt c_{553} . The charged lysine residues close to haem, which were suggested to be important for the successful promotion of electrochemical activity,⁹² are further away from the haem-TPY interface. The closest to haem charged



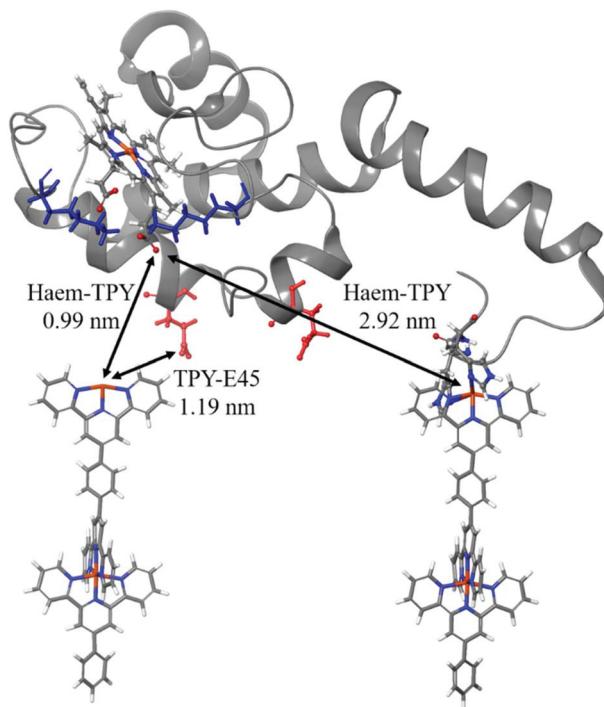


Fig. 8 Electronic-coupling pathways within the cyt *c*/TPY biomolecular system. Charged amino acids with the shortest distance to haem and TPY are depicted in blue (for lysine residues bearing the formal +1 charge) and red (for glutamic acid residues bearing the formal -1 charge), respectively. The distances were measured from the carboxylic oxygen of the haem to the iron atom of the TPY molecule.

residue (E45) is found at 1.19 nm, again further than the closest haem-TPY distance, suggesting that the ET occurs *via* a ‘through-space’ rather than ‘through-bond’ mechanism.

This hypothesis is strengthened by a relatively large distance present between the haem group and the TPY molecule connected to the cyt holoprotein *via* the 19AA peptide linker,³⁸ which is slightly below 3 nm and is 3-fold larger than the shortest haem-TPY distance. Following the protein backbone of the cyt *c*-peptide linker system, *i.e.*, from the haem to the connected TPY molecular wire, a much longer distance than 3 nm is observed. The ET pathway through the protein backbone, although cannot be excluded, is likely to yield very low ET rates, far beyond the electrochemical detection range.

Conclusions

The construction of an efficient conductive interface between electrodes and electroactive proteins is a major challenge in the biosensor and biophotovoltaics fields to achieve the desired performance of various types of nanodevices. As a major step towards this ambitious goal we report here the rational bottom-up approach to build a novel highly conductive interface based on metallo-organic TPY wires forming a uniform and well-structured SAM on the transparent ITO electrode. We show for the first time that the TPY-ligand-SAM can

be successfully used as a universal platform for specific anchoring of the electroactive protein, cyt *c*₅₅₃ on ITO. The ITO-TPY nanosystem, even without the cyt *c*₅₅₃ protein, generates photocurrents whose densities are 30-fold higher than those reported for the pristine ITO electrodes and 2-fold higher than ITO electrodes functionalised with the well-known photoactive protein PSI in the presence of an external mediator.¹⁶ Interestingly, the nature of the metal centre in the TPY wires plays a crucial role in the photocurrent generation and the overall photocurrent output. While in the case of iron the combination of Fe-TPY and cyt *c*₅₅₃ is beneficial for cathodic photocurrent generation, in the case of Co-TPY, due to BET between the haem group and the Co-TPY molecular wire, the cathodic photocurrent is diminished.

In summary, the development of a universal chemical platform for anchoring and nanostructuring of electroactive proteins reported in this study provides a major advancement for the construction of efficient (bio)molecular systems requiring a high degree of precise supramolecular organisation as well as efficient charge transfer between (photo)redox-active molecular components and various types of electrode materials to ensure maximised power output.¹⁸ Importantly, the highly conductive photoactive Fe-TPY nanosystem described here is fully based on non-toxic and Earth-abundant elements and operates in a water-based electrolyte without external mediators, which makes it attractive for future applications such as sustainable solar-to-fuel devices.

Author contributions

Margot Jacquet conceptualised and performed experiments, analysed data and wrote the original draft. Miriam Izzo expressed, purified, and biochemically/spectroscopically analysed the cyt protein and performed biofunctionalization experiments. Silvio Osella performed MD and QM/MM analyses. Sylwia Kozdra, Paweł P. Michałowski and Marian Teodorczyk provided resources and performed SIMS analysis. Dariusz Gołowicz and Krzysztof Kazimierczuk performed NMR analyses. Adam Lewera and Maciej T. Gorzkowski performed XPS analyses. Rafał Jurczakowski, Bartosz Trzaskowski and Daniel T. Gryko analysed the data and reviewed the manuscript. Joanna Kargul conceived, conceptualised and supervised all aspects of the study, co-wrote, reviewed and edited the manuscript as well as obtained funding for this work.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by funding from the National Science Centre, Poland (OPUS14 grant no. UMO-2017/27/B/ST5/00472 to J.K., HARMONIA 2016/22/M/ST5/00431 to D.T.G.



and SONATA UMO-2018/31/D/ST4/01475 to S.O.). We thank Gleb Andryianau for the initial suggestions on the organic synthesis pathways.

Notes and references

- 1 L. T. Wey, P. Bombelli, X. Chen, J. M. Lawrence, C. M. Rabideau, S. J. L. Rowden, J. Z. Zhang and C. J. Howe, *ChemElectroChem*, 2019, **6**, 5375–5386.
- 2 A. H. Teodor, B. D. Sherman, Z. Y. Ison, E. J. Ooi, J. J. Bergkamp and B. D. Bruce, *Catalysts*, 2020, **10**, 1–30.
- 3 L. Ricotti, B. Trimmer, A. W. Feinberg, R. Raman, K. K. Parker, R. Bashir, M. Sitti, S. Martel, P. Dario and A. Menciassi, *Sci. Robot.*, 2017, **2**, 1–18.
- 4 S. Arshi, M. Nozari-Asbemarz and E. Magner, *Catalysts*, 2020, **10**, 1232.
- 5 J. Tschorntner, B. Lai and J. O. Krömer, *Front. Microbiol.*, 2019, **10**, 866.
- 6 N. Kornienko, J. Z. Zhang, K. K. Sakimoto, P. Yang and E. Reisner, *Nat. Nanotechnol.*, 2018, **13**, 890–899.
- 7 P. Manickam, A. Kaushik, C. Karunakaran and S. Bhansali, *Biosens. Bioelectron.*, 2017, **87**, 654–668.
- 8 C. E. Lubner, A. M. Applegate, P. Knörzer, A. Ganago, D. A. Bryant, T. Happe and J. H. Golbeck, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 20988–20991.
- 9 J. C. Lee, H. B. Gray and J. R. Winkler, *Proc. Natl. Acad. Sci. U. S. A.*, 2001, **98**, 7760–7764.
- 10 E. F. Bowden, F. M. Hawkrige and H. N. Blount, *J. Electroanal. Chem.*, 1984, **161**, 355–376.
- 11 Z. S. Aghamiri, M. Mohsennia and H. A. Rafiee-Pour, *Talanta*, 2018, **176**, 195–207.
- 12 L. Tarpani, F. Bellezza, P. Sassi, M. Gambucci, A. Cipici and L. Latterini, *J. Phys. Chem. B*, 2019, **123**, 2567–2575.
- 13 T. Lee, S. U. Kim, J. Min and J. W. Choi, *Adv. Mater.*, 2010, **22**, 510–514.
- 14 V. M. Fribe, D. Millo, D. J. K. Swainsbury, M. R. Jones and R. N. Frese, *ACS Appl. Mater. Interfaces*, 2017, **9**, 23379–23388.
- 15 R. A. Grimme, C. E. Lubner, D. A. Bryant and J. H. Golbeck, *J. Am. Chem. Soc.*, 2008, **130**, 6308–6309.
- 16 A. Efrati, R. Tel-Vered, D. Michaeli, R. Nechushtai and I. Willner, *Energy Environ. Sci.*, 2013, **6**, 2950–2956.
- 17 K. R. Stieger, S. C. Feifel, H. Lokstein and F. Lisdat, *Phys. Chem. Chem. Phys.*, 2014, **16**, 15667–15674.
- 18 M. Kiliszek, E. Harputlu, M. Szalkowski, D. Kowalska, C. G. Unlu, P. Haniewicz, M. Abram, K. Wiwatowski, J. Niedziółka-Jönsson, S. Maćkowski, K. Ocakoglu and J. Kargul, *J. Mater. Chem. A*, 2018, **6**, 18615–18626.
- 19 M. Fedurco, *Coord. Chem. Rev.*, 2000, **209**, 263–331.
- 20 K. Nguyen and B. D. Bruce, *Biochim. Biophys. Acta, Bioenerg.*, 2014, **1837**, 1553–1566.
- 21 H. D. Sikes, J. F. Smalley, S. P. Dudek, A. R. Cook, M. D. Newton, C. E. D. Chidsey and S. W. Feldberg, *Science*, 2001, **291**, 1519–1524.
- 22 K. Slowinski, H. K. Y. Fong and M. Majda, *J. Am. Chem. Soc.*, 1999, **121**, 7257–7261.
- 23 S. Sek, A. Misicka and R. Bilewicz, *J. Phys. Chem. B*, 2000, **104**, 5399–5402.
- 24 J. J. Sumner, K. S. Weber, L. A. Hockett and S. E. Creager, *J. Phys. Chem. B*, 2000, **104**, 7449–7454.
- 25 A. Salomon, D. Cahen, S. Lindsay, J. Tomfohr, V. B. Engelkes and C. D. Frisbie, *Adv. Mater.*, 2003, **15**, 1881–1890.
- 26 J. He, F. Chen, J. Li, O. F. Sankey, Y. Terazono, C. Herrero, D. Gust, T. A. Moore, A. L. Moore and S. M. Lindsay, *J. Am. Chem. Soc.*, 2005, **127**, 1384–1385.
- 27 S. H. Choi, B. Kim and C. D. Frisbie, *Science*, 2008, **320**, 1482–1486.
- 28 G. Sedghi, K. Sawada, L. J. Esdaile, M. Hoffmann, H. L. Anderson, D. Bethell, W. Haiss, S. J. Higgins and R. J. Nichols, *J. Am. Chem. Soc.*, 2008, **130**, 8582–8583.
- 29 R. Sakamoto, K. H. Wu, R. Matsuoka, H. Maeda and H. Nishihara, *Chem. Soc. Rev.*, 2015, **44**, 7698–7714.
- 30 S. Katagiri, R. Sakamoto, H. Maeda, Y. Nishimori, T. Kurita and H. Nishihara, *Chem. – Eur. J.*, 2013, **19**, 5088–5096.
- 31 H. Maeda, R. Sakamoto and H. Nishihara, *J. Electroanal. Chem.*, 2016, **779**, 112–116.
- 32 N. Tuccitto, V. Ferri, M. Cavazzini, S. Quici, G. Zhavnerko, A. Licciardello and M. A. Rampi, *Nat. Mater.*, 2009, **8**, 41–46.
- 33 Y. Nishimori, K. Kanaizuka, T. Kurita, T. Nagatsu, Y. Segawa, F. Toshimitsu, S. Muratsugu, M. Utsuno, S. Kume, M. Murata and H. Nishihara, *Chem. – Asian J.*, 2009, **4**, 1361–1367.
- 34 H. Maeda, R. Sakamoto and H. Nishihara, *Coord. Chem. Rev.*, 2017, **346**, 139–149.
- 35 T. Wieprecht, J. Xia, U. Heinz, J. Dannacher and G. Schlingloff, *J. Mol. Catal. A: Chem.*, 2003, **203**, 113–128.
- 36 E. C. Constable and M. D. Ward, *J. Chem. Soc., Dalton Trans.*, 1990, 1405–1409.
- 37 K. T. Potts and D. Konwar, *J. Org. Chem.*, 1991, **56**, 4815–4816.
- 38 J. D. J. Olmos, P. Becquet, D. Gront, J. Sar, A. Dąbrowski, G. Gawlik, M. Teodorczyk, D. Pawlak and J. Kargul, *RSC Adv.*, 2017, **7**, 47854–47866.
- 39 J. Wang and G. S. Hanan, *Synlett*, 2005, **8**, 1251–1254.
- 40 V. Spampinato, N. Tuccitto, S. Quici, V. Calabrese, G. Marletta, A. Torrisi and A. Licciardello, *Langmuir*, 2010, **26**, 8400–8406.
- 41 C. J. Aspley and J. a. G. Williams, *New J. Chem.*, 2001, **25**, 1136–1147.
- 42 W. Goodall, K. Wild, K. J. Arm and J. A. G. Williams, *J. Chem. Soc., Perkin Trans.*, 2002, **2**, 1669–1681.
- 43 J. L. Banks, H. S. Beard, Y. Cao, A. E. Cho, W. Damm, R. Farid, A. K. Felts, T. A. Halgren, D. T. Mainz, J. R. Maple, R. Murphy, D. M. Philipp, M. P. Repasky, L. Y. Zhang, B. J. Berne, R. A. Friesner, E. Gallicchio and R. M. Levy, *J. Comput. Chem.*, 2005, **26**, 1752–1780.
- 44 A. D. Bochevarov, E. Harder, T. F. Hughes, J. R. Greenwood, D. A. Braden, D. M. Philipp, D. Rinaldo, M. D. Halls, J. Zhang and R. A. Friesner, *Int. J. Quantum Chem.*, 2013, **113**, 2110–2142.



45 A. Forget, B. Limoges and V. Balland, *Langmuir*, 2015, **31**, 1931–1940.

46 S. P. Pujari, L. Scheres, A. T. M. Marcelis and H. Zuilhof, *Angew. Chem., Int. Ed.*, 2014, **53**, 6322–6356.

47 E. Hampson, J. M. Cameron, S. Amin, J. Kyo, J. A. Watts, H. Oshio and G. N. Newton, *Angew. Chem., Int. Ed.*, 2019, **58**, 18281–18285.

48 H. Wolpher, S. Sinha, J. Pan, A. Johansson, M. J. Lundqvist, P. Persson, R. Lomoth, J. Bergquist, L. Sun, V. Sundström, B. Åkerman and T. Polívka, *Inorg. Chem.*, 2007, **46**, 638–651.

49 G. Guerrero, P. H. Mutin and A. Vioux, *Chem. Mater.*, 2001, **13**, 4367–4373.

50 A. Lanzilotto, L. A. Büldt, H. C. Schmidt, A. Prescimone, O. S. Wenger, E. C. Constable and C. E. Housecroft, *RSC Adv.*, 2016, **6**, 15370–15381.

51 M. Waser, C. Siebenhaar, J. Zampese, G. Grundler, E. Constable, M. Height and U. Pieles, *Chimia*, 2010, **64**, 328–329.

52 R. Frantz, J. O. Durand, M. Granier and G. F. Lanneau, *Tetrahedron Lett.*, 2004, **45**, 2935–2937.

53 G. Guerrero, P. H. Mutin, E. Framery and A. Vioux, *New J. Chem.*, 2008, **32**, 1519–1525.

54 S. A. Paniagua, P. J. Hotchkiss, S. C. Jones, S. R. Marder, A. Mudalige, F. S. Marrikar, J. E. Pemberton and N. R. Armstrong, *J. Phys. Chem.*, 2008, **112**, 7809–7817.

55 G. Guerrero, J. G. Alauzun, M. Granier, D. Laurencin and P. H. Mutin, *Dalton Trans.*, 2013, **42**, 12569–12585.

56 J. Lee, J. Bong, Y. G. Ha, S. Park and S. Ju, *Appl. Surf. Sci.*, 2015, **330**, 445–448.

57 M. Mennicken, S. K. Peter, C. Kaulen, U. Simon and S. Karthäuser, *J. Phys. Chem. C*, 2019, **123**, 21367–21375.

58 J. F. Moulder, W. F. Stickle, P. E. Sobol, K. D. Bomben, *Handbook of X-Ray Photoelectron Spectroscopy: A Reference Book of Standard Spectra for Identification and Interpretation of XPS Data*. Eden Prairie, MN, USA: Physical Electronics Division, Perkin-Elmer Corp, 1992.

59 B. Hirschorn, M. E. Orazem, B. Tribollet, V. Vivier, I. Frateur and M. Musiani, *Electrochim. Acta*, 2010, **55**, 6218–6227.

60 C. Xiang, Y. Zou, L. X. Sun and F. Xu, *Electrochem. Commun.*, 2008, **10**, 38–41.

61 M. Murphy, K. Theyagarajan, P. Ganesan, S. Senthilkumar and K. Thenmozhi, *Appl. Surf. Sci.*, 2019, **492**, 718–725.

62 Y. Zhao, Y. Hu, J. Hou, Z. Jia, D. Zhong, S. Zhou, D. Huo, M. Yang and C. Hou, *J. Electroanal. Chem.*, 2019, **842**, 16–23.

63 T. Pajkossy and R. Jurczakowski, *Curr. Opin. Electrochem.*, 2017, **1**, 53–58.

64 R. P. Janek, W. R. Fawcett and A. Ulman, *Langmuir*, 1998, **14**, 3011–3018.

65 M. Szalkowski, E. Harputlu, M. Kiliszek, C. G. Unlu, S. Mackowski, K. Ocakoglu, J. Kargul and D. Kowalska, *J. Mater. Chem. C*, 2020, **8**, 5807–5814.

66 R. Farran, L. Le Quang, D. Jouvenot, F. Loiseau, R. Pansu, A. Deronzier and J. Chauvin, *Inorganica Chim. Acta*, 2017, **454**, 197–207.

67 K. Kanaizuka, M. Murata, Y. Nishimori, I. Mori, K. Nishio, H. Masuda and H. Nishihara, *Chem. Lett.*, 2005, **34**, 534–535.

68 Y. Yamanoi, J. Sendo, T. Kobayashi, H. Maeda, Y. Yabusaki, M. Miyachi, R. Sakamoto and H. Nishihara, *J. Am. Chem. Soc.*, 2012, **134**, 20433–20439.

69 M. Miyachi, M. Ohta, M. Nakai, Y. Kubota, Y. Yamanoi, T. Yonezawa and H. Nishihara, *Chem. Lett.*, 2008, **37**, 404–405.

70 R. Farran, D. Jouvenot, F. Loiseau, J. Chauvin and A. Deronzier, *Dalton Trans.*, 2014, **43**, 12156–12159.

71 E. Lavoron, *J. Electroanal. Chem.*, 1979, **101**, 19–28.

72 T. Kurita, Y. Nishimori, F. Toshimitsu, S. Muratsugu, S. Kume and H. Nishihara, *J. Am. Chem. Soc.*, 2010, **132**, 4524–4525.

73 H. Maeda, R. Sakamoto and H. Nishihara, *Chem. – Eur. J.*, 2014, **20**, 2761–2764.

74 A. El Kasmi, M. C. Leopold, R. Galligan, R. T. Robertson, S. S. Saavedra, K. El Kacemi and E. F. Bowden, *Electrochem. Commun.*, 2002, **4**, 177–181.

75 Y. Wang, X. Bian, L. Liao and J. Zhu, *Microchim. Acta*, 2012, **178**, 277–283.

76 X. Jiang, L. Zhang and S. Dong, *Electrochem. Commun.*, 2006, **8**, 1137–1141.

77 N. Matsuda, H. Okabe, A. Omura, M. Nakano and K. Miyake, *Anal. Sci.*, 2017, **33**, 469–472.

78 Á. Szucs and M. Novák, *J. Electroanal. Chem.*, 1995, **384**, 47–55.

79 S. Monari, A. Ranieri, C. A. Bortolotti, S. Peressini, C. Tavagnacco and M. Borsari, *Electrochim. Acta*, 2011, **56**, 6925–6931.

80 S. Osella, M. Kiliszek, E. Harputlu, C. G. Unlu, K. Ocakoglu, B. Trzaskowski and J. Kargul, *J. Phys. Chem. C*, 2019, **123**, 8623–8632.

81 T. Cardona, A. Sedoud, N. Cox and A. W. Rutherford, *Biochim. Biophys. Acta, Bioenerg.*, 2012, **1817**, 26–43.

82 M. Y. Okamura, M. L. Paddock, M. S. Graige and G. Feher, *Biochim. Biophys. Acta, Bioenerg.*, 2000, **1458**, 148–163.

83 R. De Wijn and H. J. Van Gorkom, *Biochemistry*, 2001, **40**, 11912–11922.

84 L. M. Utschig, S. R. Greenfield, J. Tang, P. D. Laible and M. C. Thurnauer, *Biochemistry*, 1997, **36**, 8548–8558.

85 R. J. Debus, G. Feher and M. Y. Okamura, *Biochemistry*, 1986, **25**, 2276–2287.

86 S. Osella, M. Kiliszek, E. Harputlu, C. G. Unlu, K. Ocakoglu, J. Kargul and B. Trzaskowski, *J. Mater. Chem. C*, 2018, **6**, 5046–5054.

87 X. Chen, Y. C. Dai, Z. B. Zheng and K. Z. Wang, *J. Colloid Interface Sci.*, 2013, **402**, 107–113.

88 V. V. Pavlishchuk and A. W. Addison, *Inorganica Chim. Acta*, 2000, **298**, 97–102.

89 Y. Kim, K. Liang, K. Law and D. G. Whitten, *J. Phys. Chem.*, 1994, **98**, 984–988.

90 H. Imahori, H. Norieda, Y. Nishimura, I. Yamazaki, K. Higuchi, N. Kato, T. Motohiro, H. Yamada, K. Tamaki, M. Arimura and Y. Sakata, *J. Phys. Chem. B*, 2000, **104**, 1253–1260.

91 P. Salvatori, G. Marotta, A. Cinti, E. Mosconi, M. Panigrahi, L. Giribabu, M. K. Nazeeruddin and F. De Angelis, *Inorganica Chim. Acta*, 2013, **406**, 106–112.

92 P. M. Allen, H. Allen, O. Hill and N. J. Walton, *J. Electroanal. Chem. Interfacial Electrochem.*, 1984, **178**, 69–86.

