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# Anomalously polarised emission from a MoS<sub>2</sub>/WS<sub>2</sub> heterostructure†

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We report circularly polarised emission, with helicity opposite to the optical excitation, from a van der Waals heterostructure (HS) consisting of a monolayer  $MoS_2$  and three-layer  $WS_2$ . Selective excitation of the  $MoS_2$  layer confirms that this cross-polarized emission is due to the charge transfer from the  $WS_2$  layers to the  $MoS_2$  layer. We propose that the high levels of n-doping in the constituent layers due to sulphur vacancies and defects give rise to an enhanced transition rate of electrons from the k valley of  $MoS_2$ , which leads to the emission, counter polarized to the excitation. Simulations based on the rate equation model support this conclusion.

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## 1. Introduction

Two-dimensional van der Waals heterostructures (HSs) made up of atomically thin layers of transition metal dichalcogenides (TMDs) have emerged as a platform for exploring spin and valley physics and its applications. Many of these HSs have a type II band alignment, leading to efficient and ultrafast separation of the photoexcited electrons and holes into the different layers. This makes the HSs particularly interesting for light-harvesting applications. In HSs, the electrons and holes are separated in both the real and the momentum space, giving rise to a long valley lifetime. Hence, TMD HSs are promising also for valleytronics applications. The separation of the charges in different layers gives rise to interlayer excitons (IXs). There have been extensive studies focussing on the IXs well as intralayer excitons.

The HSs inherit the interesting spin-valley physics of the constituent layers and lead to numerous intriguing phenomena. Due to the large spin-orbit coupling and entangled spin and valley degrees of freedom (DoF), TMDs provide a unique platform to manipulate valley pseudospin through circularly polarized light. <sup>14,15</sup> Due to the valley contrasting Berry curvature and magnetic moment, the valley DoF can be manipulated using a magnetic field. <sup>16-19</sup> Additionally, the valley splitting can be enhanced by more than an order of magnitude by taking advantage of antiferromagnetic substrates' magnetic proximity coupling. <sup>20</sup> Doping provides another exotic way to control the exchange interactions in TMDs, leading to giant and tunable valley splitting in TMDs. TMDs grown by the chemical

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vapour deposition (CVD) method have a high density of chalcogen vacancies. In HSs with type-II band alignment, the electrons and holes get localized in different layers *via* ultrafast charge transfer.<sup>3</sup> It has been shown that this leads to a dramatic enhancement in the exchange interaction along with an anomalous enhancement in the *g*-factor.<sup>13</sup> The ultrafast charge transfer in TMD HSs also provides an all-optical route to control spin–valley phenomena by tuning the doping concentration by varying the laser power.<sup>13</sup>

Another interesting feature of TMDs, due to the valleydegenerate spin-contrasting bandstructure, is that they host both optically active and inactive intravalley excitons, called bright and dark excitons, respectively.21-24 In Mo-based monolayer TMDs, the ground state is a bright exciton, and the dark state lies energetically above it. Though most of the studies on TMDs focused on bright excitons, the existence of dark excitons has been experimentally probed via different techniques.25-27 Dark excitons are highly interesting, and many of the optical properties strongly depend on the competition between the bright and dark exciton states. Experiments have shown that the application of a magnetic field can brighten the dark exciton states.26-30 Plasmon induced brightening of dark excitons in the TMDC monolayer has also been reported.31 In doped TMD layers, trions (charged excitons) are also formed along with excitons. The application of a magnetic field has been shown to brighten the corresponding dark (charged) excitons.28 Intriguingly, the emission from the brightened dark exciton states showed helicity opposite to that of the excitation.28 However, in TMD HSs, even though IXs with helicity opposite to the excitation have been reported, reversing the helicity of intralayer exciton emission has not been reported, and intralayer excitons preserve the helicity of the excitation light. We report circularly polarised intralayer exciton emission, with helicity opposite to the optical excitation, from a MoS<sub>2</sub>/WS<sub>2</sub> heterostructure.

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Selective excitation of the  $MoS_2$  layer confirms that this cross-polarized emission is due to the charge transfer from the  $WS_2$  layers to the  $MoS_2$  layer. We propose that a dark exciton state in the  $MoS_2$  layer gets brightened due to n-doping, and the anomalous polarization arises due to the charge transfer from the conduction band of  $WS_2$  to this brightened dark exciton state. We provide simulations based on the rate equation model to support this conclusion.

## 2. Sample growth and characterisation

The WS<sub>2</sub>/MoS<sub>2</sub> HS was synthesised on a Si/SiO<sub>2</sub> substrate via single-step chemical vapour deposition (CVD) at 900 °C using a Thermo Scientific Lindberg Blue M split-tube furnace. First, WO<sub>3</sub> powder (Aldrich puriss 99.9%) is dispersed in ethanol (0.2 mg mL<sup>-1</sup>), and 10 μL dispersion is drop coated to each cleaned substrate. Similarly, MoO3 powder is dispersed in ethanol at a similar concentration. After drying the WO3 dispersion, 10 µL of MoO<sub>3</sub> dispersion is drop coated on the same substrate. Once dried in air, these substrates were loaded at the centre of the tubular furnace. As the sulfur source, 500 mg sulfur powder (Aldrich, purum 99.5%) is placed at the upstream end of the tube, just outside the heating elements. The tube is flushed several times using ultra-high-pure argon gas before the deposition to remove moisture/oxygen from the furnace. The heterostructure growth was observed at a temperature of 900 °C after a dwell time of 10 minutes while maintaining 200 sccm argon flow throughout the experiment.

Temperature-dependent and helicity resolved photoluminescence (PL) was performed using a homebuilt micro-PL setup in the backscattering configuration. For PL measurements, a 532 nm solid-state laser with an average laser power of 3.5 mW was used as the excitation source. The excitation light was focused on the sample using a 10× objective lens (NA = 0.24) which gave a spot size of  $\approx 3$  µm-diameter. The signals were collected with the same objective lens, analysed using a 0.32 m monochromator (Horiba-Jobin Yvon iHR320) and detected with a thermoelectrically cooled CCD camera. The experiments were performed at different temperatures in the range of 16-300 K using an ultralow vibration closed-cycle refrigerator. The PL and Raman maps were obtained using a Raman microscope (XploRa PLUS, Horiba Scientific) with 532 nm laser excitation. A  $50 \times$  microscope objective (NA = 0.8) was used, and it gave a spot size of  $\approx 1$  µm-diameter on the sample during PL mapping.

An atomic force microscope (AFM) image of the HS is shown in Fig. 1(a). The sample has a three-layer (3L) thick hexagonal base (WS<sub>2</sub>) with a triangular monolayer (ML) on top of it (MoS<sub>2</sub>), as indicated by the dotted lines in Fig. 1(a). The thickness of the WS<sub>2</sub> and MoS<sub>2</sub> layers is confirmed using AFM line scans, as shown in the inset of Fig. 1(a). It has been reported earlier that in WS<sub>2</sub>, compared to ML, even though the emission is weaker in thicker flakes, thicker flakes give rise to a higher degree of polarization.<sup>32,33</sup> Hence we have chosen a 3L thick WS<sub>2</sub> to form the HS. The top panel of Fig. 1(b) shows the PL from the

triangular heterostructure region, which is similar to what has been reported earlier.3,12,13,34 The PL spectra obtained from a CVD grown ML-MoS<sub>2</sub> and a 3L-WS<sub>2</sub> is also shown in Fig. 1(b) for reference. The PL spectra from HS show two peaks, an intense peak close to the charged exciton peak (AX<sup>-</sup>) in MoS<sub>2</sub> (X<sub>M</sub>) and a second peak at higher energy (X<sub>W</sub>). The dominant peak X<sub>M</sub> near the MoS<sub>2</sub> AX<sup>-</sup> peak indicates that the HS is highly doped, and the peak position agrees with the previous reports on unintentionally doped MoS<sub>2</sub>/WS<sub>2</sub> HS.<sup>13,35</sup> The X<sub>W</sub> peak is close to the emission peak observed in 3L-WS2 and the B-exciton (BX) in ML-MoS<sub>2</sub>. It has been reported that the PL from the HS is an additive spectrum of emission from the constituent WS2 and  $MoS_2$  layers in the ratio 1:5-10.<sup>5,34</sup> We believe that the  $X_W$ consists of contributions from both the exciton emission of WS<sub>2</sub> and the BX emission of  $MoS_2$ . In this paper, we focus on the  $X_M$ emission peak. Fig. 1(c) and (d) show the 2D map of the emission intensity near X<sub>M</sub> and X<sub>W</sub>. It can be seen that the emission from the triangular heterostructure region has a peak near  $X_{M}$ , and the X<sub>w</sub> emission is observed from the whole sample, including the 3L-WS<sub>2</sub> base. The Raman and PL intensity maps confirm that the MoS2 layer is present only in the triangular region, as indicated in Fig. 1(a).

Fig. 1(e) shows the Raman signals from the top of the heterostructure and the edge of the sample. The Raman signal from the top of the heterostructure consists of peaks corresponding to the in-plane  $(E_{2g}^1)$  and out-of-plane  $(A_{1g})$  modes of MoS<sub>2</sub> and WS<sub>2</sub>. The Raman signal from the edge of the sample shows only  $(E_{2g}^1)$  and  $(A_{1g})$  modes corresponding to the WS<sub>2</sub> layer, confirming that the hexagonal base consists of WS2, and MoS<sub>2</sub> is present only in the triangular region. Apart from the  $E_{2\mathrm{g}}^1$  and  $A_{1\mathrm{g}}$  modes, there are two other modes, marked U1 and U2 in ESI Fig. S1,† which appear both in the heterostructure region and the edge of the sample. However, their origin is not clear at this point. Fig. 1(f) shows the 2D map of the A<sub>1g</sub> mode of  $WS_2$ . In the triangular HS region,  $A_{1g}$  of  $WS_2$  shows a slight decrease in intensity due to the presence of the MoS2 layer. Hence, we confirm that the heterostructure consists of a hexagonal 3L-WS2 with a triangular ML-MoS2 on top of it.

## Results and discussion

We studied the polarisation characteristics of the emission from the HS. The HS was excited with a circularly polarised laser at 532 nm, and the polarisation of the PL emission was analysed. Fig. 2(a) shows the PL emission from the heterostructure at room temperature and low temperature ( $\approx$ 16 K) upon excitation with right-circularly polarised light ( $\sigma^+$ ). At room temperature, the  $X_W$  peak is predominantly co-polarised ( $\sigma^+$ ) with the excitation, while the  $X_M$  is cross-polarised ( $\sigma^-$ ) with the excitation. The peaks are blue-shifted at low temperature; however, similar polarisation dependence is observed; the  $X_M$  peak is cross-polarised ( $\sigma^-$ ), and the  $X_W$  peak is co-polarised ( $\sigma^+$ ) with the excitation. Again same polarization characteristics were observed upon excitation with left-circularly polarised light ( $\sigma^-$ ), as shown in ESI Fig. S1.†

We define the degree of circular polarisation (DCP) as  $\rho = (I_+ - I_-)/(I_+ + I_-)$ , where  $I_+$  ( $I_-$ ) denotes the intensity of the PL

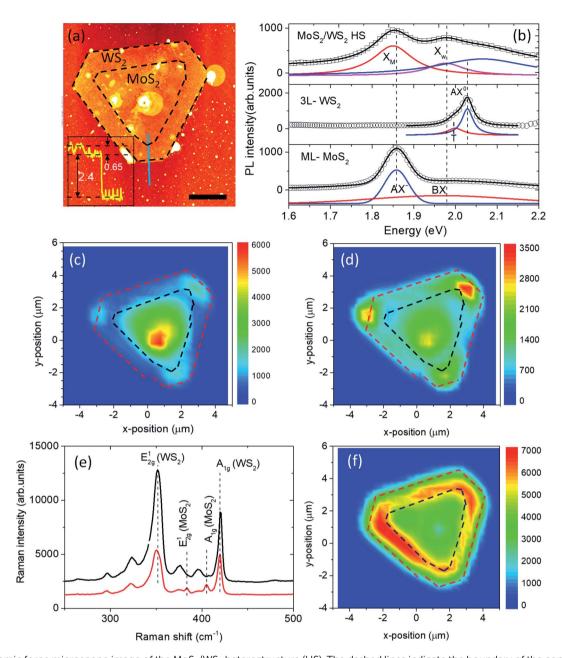


Fig. 1 (a) Atomic force microscope image of the  $MoS_2/WS_2$  heterostructure (HS). The dashed lines indicate the boundary of the constituent  $WS_2$  and  $MoS_2$  layers. The scale bar is 2 microns. Inset shows the line profile along an edge of the sample indicated by the grey line (b) low temperature (T=17~K) PL spectra from  $MoS_2/WS_2$  HS (top panel), 3L-WS<sub>2</sub> (middle panel) and ML-MoS<sub>2</sub> (bottom panel) obtained using 532 nm, linearly polarized excitation. The open symbols are the experimental data; the solid black lines are obtained by fitting the peaks to Lorentzian. The red, blue and pink lines represent the individual Lorentzian peaks. The data and the Lorentzian peaks are offset vertically for clarity. PL intensity map of (c)  $X_M$  and (d)  $X_W$  peaks in the HS sample. (e) Raman signal from the top (red) and edge (black) of the HS sample. (f) Raman intensity map of the  $A_{1g}$  peak of the HS sample.

component co-polarized (cross-polarized) with the excitation. DCP is plotted in the bottom panel of Fig. 2(a), and it can be seen that the DCP is negative near  $X_M$ . Similar measurements were performed on individual ML  $MoS_2$  and 3L  $WS_2$  flakes, shown in supplementary Fig. S2.† The ML  $MoS_2$  and the 3L  $WS_2$  flakes do not show this anomalous counter polarisation for the emission, which agree with earlier reports. 14,32,36–38 Fig. 2(b) shows the temperature dependence of the DCP near  $X_M$  and  $X_W$  peaks. For reference, the temperature dependence of the DCP of

AX $^-$  emission from ML-MoS $_2$  is also shown in Fig. 2(b). It can be seen that the DCP near the  $X_W$  peak in the HS at all temperatures, even though very low ( $\approx 1\text{-}4\%$ ), remains positive, and the DCP remains negative for the  $X_M$  peak. DCP measurements have been reported earlier on MoS $_2$ /WS $_2$  HSs, and interlayer excitons have been shown to have negative DCP. <sup>10,11,13</sup> However, intralayer exciton emission has not been reported to show negative DCP. <sup>10,13</sup> Most of the earlier work was reported on HSs comprising of MLs of MoS $_2$  and WS $_2$ . <sup>3,12,13,34</sup> Hence, we first look

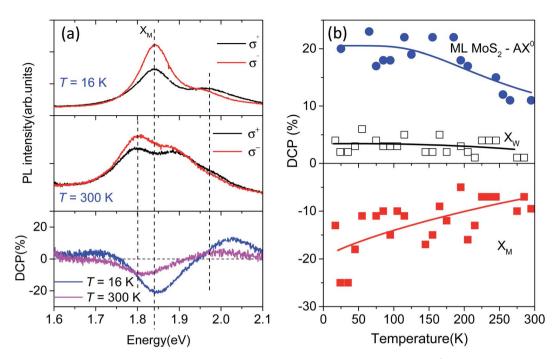


Fig. 2 (a) Helicity resolved PL spectra obtained from the MoS<sub>2</sub>/WS<sub>2</sub> HS with 532 nm, circularly polarized ( $\sigma^+$ ) excitation at T=16 K (top panel) and T = 300 K (middle panel). Degree of polarization (DCP) at T = 16 K and T = 300 K (bottom panel) obtained with 532 nm,  $\sigma^+$  excitation of the MoS<sub>2</sub>/ WS<sub>2</sub> HS. (b) DCP near the  $X_M$  and  $X_W$  peaks, at different temperatures, obtained with 532 nm,  $\sigma^+$  excitation of the MoS<sub>2</sub>/WS<sub>2</sub> HS. The DCP data corresponding to the charged exciton (AX<sup>-</sup>) in ML-MoS<sub>2</sub> are also shown in the top panel. The solid lines are a guide for the eye.

at the possibility of the negative DCP to be due to the different layer structure used here, i.e., 3L-WS<sub>2</sub> and ML-MoS<sub>2</sub>.

Even though density functional theory (DFT) often underestimates the bandgap, it can provide necessary information for distinguishing different excitations in different heterostructures. We computed the bandstructure of the HS consisting of 3L-WS2 and ML-MoS2 using DFT (see ESI Note 1† for details) and compared it with the bandstructure of the wellstudied HS comprising MLs of MoS2 and WS2 in ESI Fig. S3.† The band structure calculations demonstrate type-II band alignment. 4,5 Compared to the HS with ML-WS2, for the HS with 3L-WS<sub>2</sub>, the band minimum at the Q-point is further lowered, and the VB maximum at the  $\Gamma$ -point is further pushed up. 12,13,34 Hence the IX is at much lower energy than what has been observed in the literature and is beyond the spectral range accessible in our measurements. However, the band minima near K-points are very similar in both the HSs. Hence, the difference in the polarization behaviour of X<sub>M</sub> emission cannot be attributed to the changes in the band structure.

We first conducted measurements with a 633 nm (1.96 eV) laser excitation to understand the origin of this cross-polarised emission. 633 nm laser allowed us to selectively excite the MoS<sub>2</sub> layer in the HS and ensure that the WS2 layers were not populated. The helicity resolved PL and DCP from the HS at 17 K with 633 nm,  $\sigma^+$  polarized excitation is shown in Fig. 3. It can be seen that with 633 nm excitation, the DCP of the X<sup>M</sup> peak is positive. Similar behaviour was obtained with  $\sigma^-$  polarized 633 nm excitation as shown in ESI Fig. S4(a).† Positive DCP was obtained for all temperatures up to room temperature, with 633 nm excitation as shown in ESI Fig. S4(b).† The absence of

negative DCP when WS2 is not populated confirms that the negative DCP arises from the transfer of electrons from the WS<sub>2</sub> layer to the MoS2 layer.

After the excitation of the HS layers, charge transfer occurs, mostly via spin-conserving transfers to the lowest energy band, irrespective of the interlayer momentum mismatch.10 So the excitation of the WS<sub>2</sub> layer with  $\sigma^+$  polarized light and subsequent charge transfer could lead to the population of the spinup state in the k valley of  $MoS_2$ , which would result in the emission of  $\sigma^+$  polarized light. However, negative DCP results

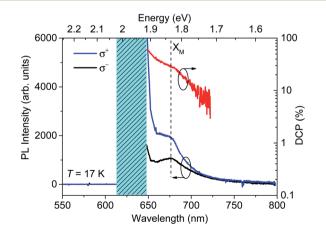


Fig. 3 Helicity resolved PL spectra and degree of polarization (DCP) obtained from the MoS<sub>2</sub>/WS<sub>2</sub> HS with 633 nm, circularly polarized ( $\sigma^+$ ) excitation at T=17 K. The shaded blue region indicates the stopband of the filter used to block the laser light from reaching the detector.

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from the dominant emission of  $\sigma^-$  polarized light. Negative DCP indicates that interlayer scattering results in the enhanced exciton population in the k' valley of MoS2, which would give rise to the emission of  $\sigma^-$  polarized light. To examine this possibility, we use a rate equation model as described below. A schematic of the band structure and the spin configuration is shown in Fig. 4(a). The system of rate equations for the  $WS_2$ MoS<sub>2</sub> HS can be written as follows:

$$\frac{\mathrm{d}N_{\mathrm{w}}}{\mathrm{d}t} = g - \frac{N_{\mathrm{w}}}{\tau_{\mathrm{r}}^{\mathrm{w}}} - \frac{N_{\mathrm{w}}}{\tau} - \frac{N_{\mathrm{w}}}{\tau'}$$

$$\frac{dN_{\text{Mo}}}{dt} = g - \frac{N_{\text{Mo}}}{\tau_{\text{r}}^{\text{Mo}}} + \frac{N_{\text{w}}}{\tau} - \frac{N_{\text{Mo}}}{\tau_{\text{s}}^{\text{Mo}}} - \frac{N'_{\text{Mo}}}{\tau_{\text{s}}^{\text{Mo}}}$$

$$\frac{dN'_{Mo}}{dt} = g' - \frac{N'_{Mo}}{\tau_{r}^{Mo}} + \frac{N_{w}}{\tau'} + \frac{N_{Mo}}{\tau_{s}^{Mo}} - \frac{N'_{Mo}}{\tau_{s}^{Mo}}$$

 $N_{\text{Mo}}(N'_{\text{Mo}})$  is the exciton population in the k(k') valley of the  $MoS_2$  layer and  $N_w$  the exciton population in the k valley of the WS<sub>2</sub> layer. g(g') is the generation rate in the k(k') valley, which is assumed to be a constant for continuous-wave excitation. We assumed that  $\sigma^+$  excitation generates excitons in the k valleys of MoS<sub>2</sub> and WS<sub>2</sub> layers, and for simplicity, we have ignored the scattering of electrons to the k' valley of the WS2 layer. This assumption is valid since, due to the type-II band alignment, the transfer of photogenerated electrons from the WS2 k valley to the MoS<sub>2</sub> valleys is more efficient than their scattering to the k'valley of the WS<sub>2</sub>.<sup>39</sup>  $\tau_s^{Mo}$  and  $\tau_r^{Mo}$  represent the intervalley scattering time and the radiative recombination time, respectively, in the MoS<sub>2</sub> layer.  $\tau_r^w$  represents the radiative recombination time in the WS<sub>2</sub> layer.  $\tau$  ( $\tau'$ ) represents the intralayer scattering time between the k valley in WS<sub>2</sub> and the k(k') valley in MoS<sub>2</sub>. The polarization state is calculated under steady-state condi-

tions  $\left(\frac{dN_{\rm w}}{dt} = \frac{dN_{\rm Mo}}{dt} = \frac{dN'_{\rm Mo}}{dt} = 0\right)$ . The DCP of the  $X_{\rm M}$  emission is related to the population of exciton states in the k(k')valleys of MoS<sub>2</sub>,

$$ho = rac{N_{ ext{Mo}} - N^{\prime}_{ ext{Mo}}}{N_{ ext{Mo}} + N^{\prime}_{ ext{Mo}}}$$

Solving the above set of rate equations for g = 1 and g' = 0 we obtain

$$\rho = \frac{\gamma_{\rm r}^{\rm Mo}(\gamma(1-\alpha)+1)}{\left(2+\gamma_{\rm r}^{\rm Mo}\right)(\gamma(1+\alpha)+1)}$$

 $\rho = \frac{\gamma_r^{Mo}(\gamma(1-\alpha)+1)}{\left(2+\gamma_r^{Mo}\right)(\gamma(1+\alpha)+1)}$  where  $\gamma_r^{Mo} = \frac{\tau_s^{Mo}}{\tau_r^{Mo}}$  and  $\gamma = \frac{\tau_s^{Mo}}{\tau}$  represent the relaxation rates and  $\alpha = \frac{\tau}{\tau'}$ .

If we ignore the interlayer scattering ( $\gamma = 0$ ), the above equation describes an isolated MoS2 layer, and it reduces to  $\rho = \frac{1}{1 + \frac{2}{\gamma_r^{Mo}}} = \frac{1}{1 + 2\frac{\tau_r^{Mo}}{\tau_r^{Mo}}}.$  In that case, we obtain  $\rho \approx 20\%$  for

 $\gamma_{\rm r}^{\rm Mo} \approx 1/2$ , which agrees with the  $\rho$  obtained for ML-MoS<sub>2</sub> at low temperature.  $\tau_r^{\text{Mo/W}}$  is in the range of picoseconds to hundreds picoseconds.<sup>25,41,42</sup> The intervalley scattering time  $\tau_{\rm s}^{\rm Mo}$  depends on the excitation energy but will be of the same order of  $\tau_r^{\text{Mo/W}}$ . It has been shown that the interlayer scattering time  $(\tau', \tau)$  is of the order of 50–450 fs (ref. 3, 34, 43 and 44) which means  $\gamma$ ,  $\gamma' > 1$ . Fig. 4(b) shows  $\rho$  as a function of  $\alpha$  for  $\gamma_{\rm r}^{\rm Mo} = 1/2$  and for different values of  $\gamma$ , which shows that negative DCP is indeed possible for  $\tau' < \tau$ . This means that electrons photoexcited to the k valley of WS<sub>2</sub> by  $\sigma^+$  polarised light are scattered preferentially to the k' valley compared to the k valley of MoS<sub>2</sub>. We believe that this is due to unintentional doping of the HS. CVD grown samples have a high density of sulphur vacancies, leading to high levels of n-doping which is evident from the emission X<sub>M</sub> near the charged exciton energy in MoS2.13 The high levels of n-doping has two effects. Firstly, the Fermi level will be located in the spin-split upper conduction band of MoS2.13 Because of this, the electrons from WS2 layers will be scattered to the spin-split upper conduction bands of MoS<sub>2</sub>. The transition to the spin-split upper conduction band of  $MoS_2$  k' valley  $(m_i = -3/2)$  would be a spin-conserving transition. Hence, possibly, the transition rate of this spinconserving transition is enhanced compared to the spin-flip

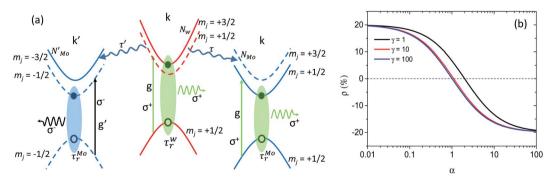


Fig. 4 (a) A schematic of the different bands in the MoS<sub>2</sub>/WS<sub>2</sub> HS. Solid and dotted lines represent bands with different spin configurations. The bands are labelled with the z-component of their total angular momentum, m<sub>i</sub>. Vertical arrows indicate optical excitations: black and green arrows correspond to  $\sigma^-$  and  $\sigma^+$  excitation, respectively. The black and green wiggly arrows indicate  $\sigma^-$  and  $\sigma^+$  emission, respectively. The gray wiggly arrows represent spin-conserving interlayer transfer to the lowest energy band. (b) Degree of circular polarization,  $\rho$ , calculated using the rate equation model described in the main text for  $\gamma_r^{Mo}=1/2$  and different values of  $\alpha$ .

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transition to the k valley ( $m_i = +3/2$ ). Normally, the spinconserving transition from WS<sub>2</sub> to the spin-split upper conduction band of the MoS2 k' valley would give rise to dark (neutral or charged) excitons. However, as another consequence of the n-doping of the layers the dark exciton state is brightened, similar to the brightening observed earlier for the dark IX state in MoS<sub>2</sub>/WS<sub>2</sub> HSs due to electrostatic doping.9

#### 4. Conclusions

In conclusion, we unveil anomalous polarization from a van der Waals heterostructure (HS) consisting of ML-MoS<sub>2</sub> and 3L-WS<sub>2</sub>. Selective excitation of the MoS<sub>2</sub> layer in the HS demonstrates that this anomalous polarization originates from the charge transfer from WS<sub>2</sub> to MoS<sub>2</sub> layers. We propose that due to high levels of n-doping in the layers, caused by sulphur vacancies and defects, the transition rate of electrons from the k valley of WS<sub>2</sub> to the k' valley of MoS<sub>2</sub> is enhanced, which leads to emission, counter polarized ( $\sigma^-$ ) to the excitation ( $\sigma^+$ ). We simulated this scenario using a set of coupled rate equations, and the simulations agree well with our experimental observations. Our measurements provide new prospects to control the helicity of the emission from van der Waals HSs via doping.

## Author contributions

PRM and PKB did the measurements and analysed the data. PVS grew the samples. ASK did the DFT calculations. SS did the simulations based on the rate equation model. MMS designed and supervised the growth of nanostructures. RNK supervised the measurements & analysis and prepared the manuscript with inputs from all co-authors.

### Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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## References

- 1 A. K. Geim and I. V. Grigorieva, Van der Waals heterostructures, Nature, 2013, 499, 419-425.
- 2 P. Rivera, H. Yu, K. L. Seyler, N. P. Wilson, W. Yao and X. Xu, Interlayer valley excitons in heterobilayers of transition metal dichalcogenides, Nat. Nanotechnol., 2018, 13, 1004-1015.

- 3 X. Hong, J. Kim, S. F. Shi, Y. Zhang, C. Jin, Y. Sun, S. Tongay, J. Wu, Y. Zhang and F. Wang, Ultrafast charge transfer in atomically thin MoS<sub>2</sub>/WS<sub>2</sub> heterostructures, Nanotechnol., 2014, 9, 682-686.
- 4 J. Kang, S. Tongay, J. Zhou, J. Li and J. Wu, Band offsets and heterostructures of two-dimensional semiconductors, Appl. Phys. Lett., 2013, 102, 12111.
- 5 H. P. Komsa and A. V. Krasheninnikov, Electronic structures and optical properties of realistic transition metal dichalcogenide heterostructures from first principles, Phys. Rev. B: Condens. Matter Mater. Phys., 2013, 88, 085318.
- 6 J. Kim, C. Jin, B. Chen, H. Cai, T. Zhao, P. Lee, S. Kahn, K. Watanabe, T. Taniguchi, S. Tongay, M. F. Crommie and F. Wang, Observation of ultralong valley lifetime in WSe<sub>2</sub>/ MoS<sub>2</sub> heterostructures, Sci. Adv., 2017, 3, e1700518.
- 7 C. Jin, J. Kim, M. Iqbal Bakti Utama, E. C. Regan, H. Kleemann, H. Cai, Y. Shen, M. J. Shinner, A. Sengupta, K. Watanabe, T. Taniguchi, S. Tongay, A. Zettl and F. Wang, Imaging of pure spin-valley diffusion current in WS<sub>2</sub>-WSe<sub>2</sub> heterostructures, Science, 2018, 360, 893–896.
- 8 P. Rivera, K. L. Seyler, H. Yu, J. R. Schaibley, J. Yan, D. G. Mandrus, W. Yao and X. Xu, Valley-polarized exciton dynamics in a 2D semiconductor heterostructure, Science, 2016, 351, 688-691.
- 9 A. Ciarrocchi, D. Unuchek, A. Avsar, K. Watanabe, T. Taniguchi and A. Kis, Polarization switching and electrical control of interlayer excitons in two-dimensional van der Waals heterostructures, Nat. Photonics, 2019, 13, 131-136.
- 10 W. T. Hsu, L. S. Lu, P. H. Wu, M. H. Lee, P. J. Chen, P. Y. Wu, Y. C. Chou, H. T. Jeng, L. J. Li, M. W. Chu and W. H. Chang, Negative circular polarization emissions from WSe<sub>2</sub>/MoSe<sub>2</sub> commensurate heterobilayers, Nat. Commun., 2018, 9, 1356.
- 11 A. T. Hanbicki, H. J. Chuang, M. R. Rosenberger, C. S. Hellberg, S. V. Sivaram, K. M. McCreary, I. I. Mazin and B. T. Jonker, Double Indirect Interlayer Exciton in a MoSe<sub>2</sub>/WSe<sub>2</sub> van der Waals Heterostructure, ACS Nano, 2018, 12, 4719-4726.
- 12 M. Okada, A. Kutana, Y. Kureishi, Y. Kobayashi, Y. Saito, T. Saito, K. Watanabe, T. Taniguchi, S. Gupta, Y. Miyata, B. I. Yakobson, H. Shinohara and R. Kitaura, Direct and Indirect Interlayer Excitons in a van der Waals Heterostructure of hBN/WS2/MoS2/hBN, ACS Nano, 2018, 12, 2498-2505.
- 13 J. Zhang, L. Du, S. Feng, R. W. Zhang, B. Cao, C. Zou, Y. Chen, M. Liao, B. Zhang, S. A. Yang, G. Zhang and T. Yu, Enhancing and controlling valley magnetic response in MoS<sub>2</sub>/WS<sub>2</sub> heterostructures by all-optical route, Nat. Commun., 2019, 10, 4226.
- 14 K. F. Mak, K. He, J. Shan and T. F. Heinz, Control of valley polarization in monolayer MoS<sub>2</sub> by optical helicity, Nat. Nanotechnol., 2012, 7, 494-498.
- 15 H. Zeng, J. Dai, W. Yao, D. Xiao and X. Cui, Valley polarization in MoS<sub>2</sub> monolayers by optical pumping, Nat. Nanotechnol., 2012, 7, 490-493.
- 16 Y. Li, J. Ludwig, T. Low, A. Chernikov, X. Cui, G. Arefe, Y. D. Kim, A. M. Van Der Zande, A. Rigosi, H. M. Hill,

- S. H. Kim, J. Hone, Z. Li, D. Smirnov and T. F. Heinz, Valley splitting and polarization by the Zeeman effect in monolayer MoSe<sub>2</sub>, *Phys. Rev. Lett.*, 2014, **113**, 1–5.
- 17 A. Srivastava, M. Sidler, A. V. Allain, D. S. Lembke, A. Kis and A. Imamoğlu, Valley Zeeman effect in elementary optical excitations of monolayer WSe<sub>2</sub>, *Nat. Phys.*, 2015, **11**, 141–147.
- 18 G. Aivazian, Z. Gong, A. M. Jones, R. L. Chu, J. Yan, D. G. Mandrus, C. Zhang, D. Cobden, W. Yao and X. Xu, Magnetic control of valley pseudospin in monolayer WSe<sub>2</sub>, *Nat. Phys.*, 2015, 11, 148–152.
- 19 D. Macneill, C. Heikes, K. F. Mak, Z. Anderson, A. Kormányos, V. Zólyomi, J. Park and D. C. Ralph, Breaking of valley degeneracy by magnetic field in monolayer MoSe<sub>2</sub>, *Phys. Rev. Lett.*, 2015, 114, 1–5.
- 20 L. Xu, M. Yang, L. Shen, J. Zhou, T. Zhu and Y. P. Feng, Large valley splitting in monolayer WS<sub>2</sub> by proximity coupling to an insulating antiferromagnetic substrate, *Phys. Rev. B*, 2018, 97, 2–7.
- 21 H. Yu, G. B. Liu, P. Gong, X. Xu and W. Yao, Dirac cones and Dirac saddle points of bright excitons in monolayer transition metal dichalcogenides, *Nat. Commun.*, 2014, 5, 1–7.
- 22 H. Yu, X. Cui, X. Xu and W. Yao, Valley excitons in twodimensional semiconductors, *Natl. Sci. Rev.*, 2015, 2, 57–70.
- 23 C. Robert, D. Lagarde, F. Cadiz, G. Wang, B. Lassagne, T. Amand, A. Balocchi, P. Renucci, S. Tongay, B. Urbaszek and X. Marie, Exciton radiative lifetime in transition metal dichalcogenide monolayers, *Phys. Rev. B*, 2016, 93, 1–10.
- 24 G. Berghäuser, P. Steinleitner, P. Merkl, R. Huber, A. Knorr and E. Malic, Mapping of the dark exciton landscape in transition metal dichalcogenides, *Phys. Rev. B*, 2018, **98**, 1–6.
- 25 X. X. Zhang, Y. You, S. Y. F. Zhao and T. F. Heinz, Experimental Evidence for Dark Excitons in Monolayer WSe<sub>2</sub>, *Phys. Rev. Lett.*, 2015, **115**, 1–6.
- 26 Z. Li, T. Wang, C. Jin, Z. Lu, Z. Lian, Y. Meng, M. Blei, S. Gao, T. Taniguchi, K. Watanabe, T. Ren, S. Tongay, L. Yang, D. Smirnov, T. Cao and S. F. Shi, Emerging photoluminescence from the dark-exciton phonon replica in monolayer WSe<sub>2</sub>, Nat. Commun., 2019, 10, 1–7.
- 27 M. R. Molas, C. Faugeras, A. O. Slobodeniuk, K. Nogajewski, M. Bartos, D. M. Basko and M. Potemski, Brightening of dark excitons in monolayers of semiconducting transition metal dichalcogenides, 2D Mater., 2017, 4, 021003.
- 28 X. X. Zhang, T. Cao, Z. Lu, Y. C. Lin, F. Zhang, Y. Wang, Z. Li, J. C. Hone, J. A. Robinson, D. Smirnov, S. G. Louie and T. F. Heinz, Magnetic brightening and control of dark excitons in monolayer WSe<sub>2</sub>, *Nat. Nanotechnol.*, 2017, 12, 883–888.
- 29 R. Vasconcelos, H. Bragança, F. Qu and J. Fu, Dark exciton brightening and its engaged valley dynamics in monolayer WSe<sub>2</sub>, *Phys. Rev. B*, 2018, **98**, 195302.
- 30 M. Feierabend, S. Brem, A. Ekman and E. Malic, Brightening of spin- and momentum-dark excitons in transition metal dichalcogenides, *2D Mater.*, 2021, **8**, 015013.
- 31 A. Arora, T. Dixit, K. V. Anil Kumar, S. Krishnan, K. Lakshmi Ganapathi, A. Krishnan, P. K. Nayak and M. S. Ramachandra

- Rao, Plasmon induced brightening of dark exciton in monolayer WSe<sub>2</sub> for quantum optoelectronics, *Appl. Phys. Lett.*, 2019, **114**, 201101.
- 32 L. Du, J. Tang, J. Liang, M. Liao, Z. Jia, Q. Zhang, Y. Zhao, R. Yang, D. Shi, L. Gu, J. Xiang, K. Liu, Z. Sun and G. Zhang, Giant Valley Coherence at Room Temperature in 3R WS<sub>2</sub> with Broken Inversion Symmetry, *Research*, 2019, 2019, 1–8.
- 33 P. K. Barman, P. V. Sarma, M. M. Shaijumon and R. N. Kini, High degree of circular polarization in WS<sub>2</sub> spiral nanostructures induced by broken symmetry, *Sci. Rep.*, 2019, **9**, 1–7.
- 34 S. Tongay, W. Fan, J. Kang, J. Park, U. Koldemir, J. Suh, D. S. Narang, K. Liu, J. Ji, J. Li, R. Sinclair and J. Wu, Tuning interlayer coupling in large-area heterostructures with CVD-grown MoS<sub>2</sub> and WS<sub>2</sub> monolayers, *Nano Lett.*, 2014, 14, 3185–3190.
- 35 K. F. Mak, K. He, C. Lee, G. H. Lee, J. Hone, T. F. Heinz and J. Shan, Tightly bound trions in monolayer MoS<sub>2</sub>, *Nat. Mater.*, 2013, **12**, 207–211.
- 36 H. Zeng, J. Dai, W. Yao, D. Xiao and X. Cui, Valley polarization in MoS<sub>2</sub> monolayers by optical pumping, *Nat. Nanotechnol.*, 2012, 7, 490–493.
- 37 B. Zhu, H. Zeng, J. Dai, Z. Gong and X. Cui, Anomalously robust valley polarization and valley coherence in bilayer WS<sub>2</sub>, *Proc. Natl. Acad. Sci. U. S. A.*, 2014, **111**, 11606–11611.
- 38 P. K. Nayak, F. C. Lin, C. H. Yeh, J. S. Huang and P. W. Chiu, Robust room temperature valley polarization in monolayer and bilayer WS<sub>2</sub>, *Nanoscale*, 2016, **8**, 6035–6042.
- 39 Z. Li, G. Ezhilarasu, I. Chatzakis, R. Dhall, C. C. Chen and S. B. Cronin, Indirect Band Gap Emission by Hot Electron Injection in Metal/MoS<sub>2</sub> and Metal/WSe<sub>2</sub> Heterojunctions, *Nano Lett.*, 2015, 15, 3977–3982.
- 40 G. Kioseoglou, A. T. Hanbicki, M. Currie, A. L. Friedman, D. Gunlycke and B. T. Jonker, Valley polarization and intervalley scattering in monolayer MoS<sub>2</sub>, *Appl. Phys. Lett.*, 2012, 101, 221907.
- 41 T. Korn, S. Heydrich, M. Hirmer, J. Schmutzler and C. Schller, Low-temperature photocarrier dynamics in monolayer MoS<sub>2</sub>, *Appl. Phys. Lett.*, 2011, **99**, 2–5.
- 42 M. Amani, D. H. Lien, D. Kiriya, J. Xiao, A. Azcatl, J. Noh, S. R. Madhvapathy, R. Addou, K. C. Santosh, M. Dubey, K. Cho, R. M. Wallace, S. C. Lee, J. H. He, J. W. Ager, X. Zhang, E. Yablonovitch and A. Javey, Near-unity photoluminescence quantum yield in MoS<sub>2</sub>, *Science*, 2015, 350, 1065–1068.
- 43 K. Wang, B. Huang, M. Tian, F. Ceballos, M. W. Lin, M. Mahjouri-Samani, A. Boulesbaa, A. A. Puretzky, C. M. Rouleau, M. Yoon, H. Zhao, K. Xiao, G. Duscher and D. B. Geohegan, Interlayer Coupling in Twisted WSe<sub>2</sub>/WS<sub>2</sub> Bilayer Heterostructures Revealed by Optical Spectroscopy, ACS Nano, 2016, 10, 6612–6622.
- 44 F. Ceballos, M. Z. Bellus, H. Y. Chiu and H. Zhao, Ultrafast charge separation and indirect exciton formation in a MoS<sub>2</sub>-MoSe<sub>2</sub> van der waals heterostructure, *ACS Nano*, 2014, 8, 12717–12724.