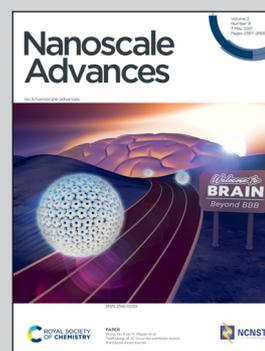


Showcasing research from Professor F. Lagugné-Labarthet's laboratory, Department of Chemistry, Western University, London, Ontario, Canada in collaboration with Professors N. Félidj and C. Mangeney, Université de Paris.

Three-color plasmon-mediated reduction of diazonium salts over metasurfaces

Metastructures that exhibit multiple plasmon resonances can be utilized to spatially control surface functionalization using polarization state and resonant wavelengths. This work highlights how three distinct wavelengths are used to spatially pattern diazonium salts in distinct fashions in the vicinity of individual gold structures. This opens future applications in sensing, colour control and surface chemistry where multiplexed patterning of distinct molecules around a given nanostructure plays a key role.

As featured in:



See François Lagugné-Labarthet *et al.*, *Nanoscale Adv.*, 2021, 3, 2501.

Cite this: *Nanoscale Adv.*, 2021, 3, 2501

## Three-color plasmon-mediated reduction of diazonium salts over metasurfaces†

Denis A. B. Therien,<sup>a</sup> Danielle M. McRae,<sup>a</sup> Claire Mangeney,<sup>id</sup><sup>b</sup> Nordin Félidj<sup>id</sup><sup>c</sup> and François Lagugné-Labarhet<sup>id</sup><sup>\*a</sup>

Surface plasmon-mediated chemical reactions are of great interest for a variety of applications ranging from micro- and nanoscale device fabrication to chemical reactions of societal interest for hydrogen production or carbon reduction. In this work, a crosshair-like nanostructure is investigated due to its ability to induce local enhancement of the local electromagnetic field at three distinct wavelengths corresponding to three plasmon resonances. The structures are irradiated in the presence of a solution containing diazonium salts at wavelengths that match the resonance positions at 532 nm, 632.8 nm, and 800 nm. The resulting grafting shows polarization and wavelength-dependent growth patterns at the nanoscale. The plasmon-mediated reactions over arrays of the crosshair structures are further investigated using scanning electron microscopy and supported by finite domain time domain modelling revealing wavelength and polarization specific reactions. Such an approach enables nanoscale molecular printing using light source opening multiplexing applications where different analytes can be grafted under distinct opto-geometric conditions.

Received 14th October 2020  
Accepted 20th February 2021

DOI: 10.1039/d0na00862a

rsc.li/nanoscale-advances

## Introduction

Excitation of localized surface plasmon resonances (LSPRs) is an effective method for enhancing the local optical fields which can in turn be used for a variety of applications such as in enhanced spectroscopies and plasmonic sensing where both sensitivity and spatial resolution can be drastically improved through a plasmon effect.<sup>1–4</sup> In discrete nanostructures, LSPRs are generally spatially confined in specific areas and can be excited with a selected set of wavelengths and polarization of the impinging field. In ideal opto-geometric configuration, the electric field is typically enhanced by a 10–100 fold factor which is enough to yield large enhancement factors and which can in turn be used for a variety of sensing applications through surface enhanced mechanisms in Raman,<sup>2,3,5–7</sup> infrared,<sup>8–10</sup> fluorescence,<sup>11–14</sup> and nonlinear optical measurements.<sup>15–18</sup>

This spatial confinement can also be used to promote chemical reactions through the generation and ejection of hot electrons.<sup>19</sup> Metallic nanomaterials that contain LSPRs have shown wide applicability for plasmon-mediated reactions.<sup>20–22</sup>

Such reactions include chain-linking of gold nanoparticles,<sup>23</sup> fabrication of biosensors,<sup>24</sup> plasmon mediated drilling,<sup>22</sup> and performing surface chemistry such as CO<sub>2</sub> reduction,<sup>25–27</sup> water splitting for hydrogen production,<sup>28–30</sup> single oxygen production,<sup>14</sup> and artificial photosynthesis.<sup>31</sup> Multiplexed functionalization is of particular interest when several analytes can be spatially grafted over metallic structures solely using light irradiation. Such reactions have been studied on gold nanodisks by grafting both a carboxyphenyl diazonium salt and a hydroxyethyl phenyl diazonium salt to a single surface using orthogonal irradiation polarizations.<sup>32</sup> A polymer formulation containing two different light-emitting quantum dots was grafted to the surface of a nanomaterial at two different polarizations, highlighting the possibility of grafting quantum dots in orthogonal directions keeping at a nanoscale resolution.<sup>33</sup> Other plasmon-mediated reactions include local polymerization in the vicinity of metallic nanostructures and temperature-responsive gold/polymer hybrid structures.<sup>34,35</sup> Such light-induced nanoscale printing opens many perspectives in device fabrication. The main limitation in multi-analyte functionalization is the number of plasmon resonances that can be tackled independently with different wavelengths. A nanostructure with *n* discrete resonances will theoretically open the possibility of using *n* wavelengths of irradiations. Each resonance being associated with a unique spatial pattern of the electromagnetic confinement will subsequently yield a defined pattern upon irradiation with the matching wavelength. Such structures with multiple resonances have been demonstrated in fractal structures with resonances ranging from the visible to

<sup>a</sup>Department of Chemistry, University of Western Ontario, 1151 Richmond Street, London, ON, N6A 5B7, Canada. E-mail: flagugne@uwo.ca

<sup>b</sup>Université de Paris, Laboratoire de Chimie et Biochimie Pharmacologiques et Toxicologiques, LCBPT, UMR 8601 CNRS, 45 rue des Saints Pères, F-75006 Paris, France

<sup>c</sup>Université de Paris, ITODYS, UMR 7086 CNRS, 15 rue J.-A. de Baïf, F-75013 Paris, France

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0na00862a



the mid-infrared and highlighting the hybridization between the successive fractal generations.<sup>3,21,36</sup>

In the present work, we focus on arrays of gold meta-structures with crosshair geometries and that display three resonances in the visible range. The nanostructures were made by electron beam lithography and display three resonances centered around 532, 632 and 785 nm. Specifically, we investigate the spatial grafting of 4-nitrobenzenediazonium tetrafluoroborate that has shown excellent reactivity when subjected to light irradiation in the vicinity of metallic nanostructures and forms a thin polymer film that can be readily observed by electron microscopy.

## Materials and methods

### Finite domain time domain (FDTD) modelling

Simulations throughout this study were performed using Lumerical FDTD solutions (Vancouver, Canada). The properties of the structures and substrate simulated were set using the standard Palik titanium and gold indices of refraction.<sup>37</sup> The structures were inscribed on a glass coverslip. The symmetric structures were inscribed using a thickness of 3 nm for the titanium adhesion layer and 20 nm for the gold. An array of structures was set up with an average of 800 nm between individual structures in both the *x*- and *y*-directions, with periodic boundary conditions, and perfectly matched layers along the *z*-axis. In order to reduce the computation time, the symmetry of the structure was exploited by applying symmetry and anti-symmetry boundary conditions appropriately. The amplitude of the input planar wave field was set to 1 V m<sup>-1</sup> for the simulations, and an input light source with a wavelength range of 400 to 1000 nm was used. The near-field maps were calculated at the surface of the nanostructure at wavelengths of 530, 630, and 800 nm meanwhile the scattering spectrum was calculated in the far-field.

### Electron beam lithography (EBL)

EBL was used to prepare the crosshair nanostructures in arrays on (22 × 22) mm<sup>2</sup> glass coverslips (Thermo Fisher Scientific, ON, CA). The glass coverslips were cleaned using an O<sub>2</sub> plasma cleaner. Following this, poly(methyl methacrylate) in anisole (PMMA A2 950A resist, MicroChem Corp. MA, USA) was spin coated onto glass slides at 3000 rpm, corresponding to a thickness of 200 nm. Then, a conductive polymer layer (AquaSAVE, Mitsubishi Rayon America) was spin-coated over the resist to prevent loss of resolution during lithography due to charging effects. The samples were then loaded into a scanning electron microscope (LEO 1530) and EBL was performed using the ELPHY Quantum nanolithography and nanofabrication software from Raith Nanofabrication. Following irradiation, the samples were developed in methyl isobutyl ketone : isopropanol (MIBK : IPA, 1 : 3 [v : v], Microchem Corp., MA, USA) for three minutes to generate the mask of the structures. Subsequently, the samples were immersed in isopropanol (VWR International LLC., ON, CA) for three minutes to stop the development process. Metal deposition was performed using an

electron beam evaporation instrument (Angstrom Engineering). A 3 nm titanium adhesion layer was added at a rate of 0.8 Å s<sup>-1</sup> to ensure that the gold would adhere to the surface of the glass, followed by 20 nm of gold at a rate of 1 Å s<sup>-1</sup>. Finally, a lift-off procedure in acetone was used to remove the mask and excess metal, leaving the desired arrays.

### Diazonium salt grafting

A 5 mM solution of 4-nitrobenzenediazonium tetrafluoroborate was prepared in Milli-Q water (18.2 Ω) and drop-casted over the surface of the platforms positioned onto an inverted microscope. A rubber O-ring was used to keep the solution on top of the array during irradiation and to prevent the samples from drying. After irradiation, the samples were washed with copious amounts of Milli-Q water and dried under a gentle flow of nitrogen gas.

Irradiation was performed using CW wavelengths at 532 nm and 632.8 nm. Irradiation at 800 nm was performed with a femtosecond laser (pulse width ~ 200 fs, repetition rate 80 MHz). The beam size was approximately 1 μm for the 800 nm irradiation and about 8 μm for the 532 nm and 632.8 nm lasers. The laser irradiance at the sample plane was 1.79 × 10<sup>3</sup> W cm<sup>-2</sup> for 532 nm, 1.17 × 10<sup>4</sup> W cm<sup>-2</sup> for 632.8 nm, and 1.27 × 10<sup>6</sup> W cm<sup>-2</sup> for 800 nm.

### Scanning electron microscopy (SEM)

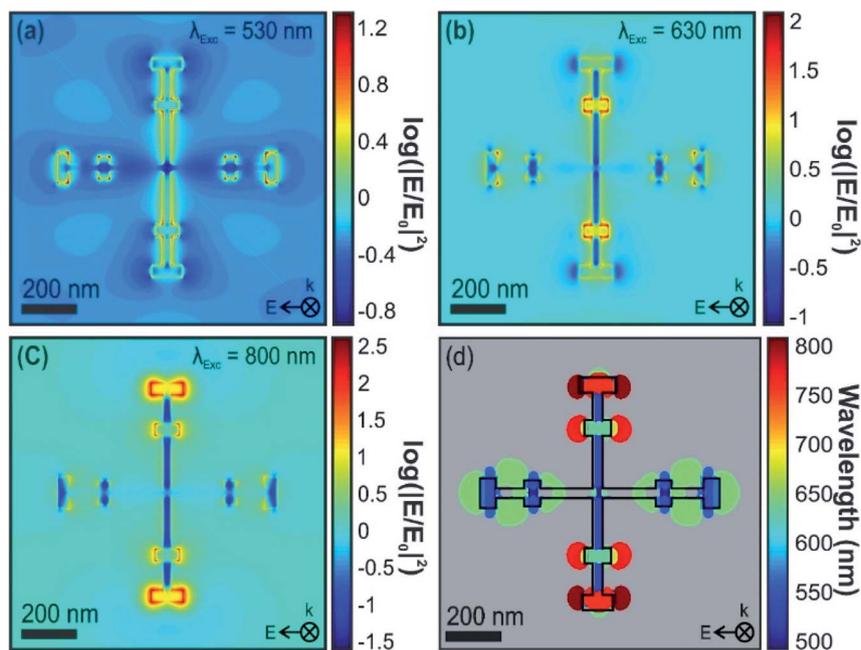
SEM was performed using a LEO (Zeiss) 1530 field emission SEM. Since all samples were inscribed on glass coverslips, samples were coated with 5 nm of osmium prior to SEM imaging to prevent charging on the surface.

## Results and discussion

Crosshair-like nanostructures, dendrimer-like structures with *C*<sub>4</sub> symmetry, were first designed and simulated using FDTD calculations, allowing their plasmon resonances to be fine-tuned prior to fabrication.

The near-field distribution of the EM field around the structure has been calculated at 530 nm, 630 nm and 800 nm for a horizontal polarization over the crosshair structures that are composed of orthogonal major rods with a length of 800 nm and with two minor rods across each arm of the structure. Irradiation at 530 nm (Fig. 1a) shows confinement of the field along all rods, major and minor, that are orthogonal to the polarization. At 630 nm (Fig. 1b), the two minor rods located in the middle of the branch orthogonal to the input polarization display field confinement while at 800 nm (Fig. 1c) the two minor rods located at the extremity of the branch on the arms perpendicular to the polarization direction concentrate the field the most. The isowavelength map shown in Fig. 1d provides a snapshot of the different zones where the field is locally enhanced for the three selected wavelengths. The dark blue, green and red colors indicate the field confinement at 532 nm, 632 nm and 800 nm, respectively. The structure being isotropic with a *C*<sub>4</sub> symmetry axis, a rotation of the polarization by 90° will rotate the distribution pattern accordingly.



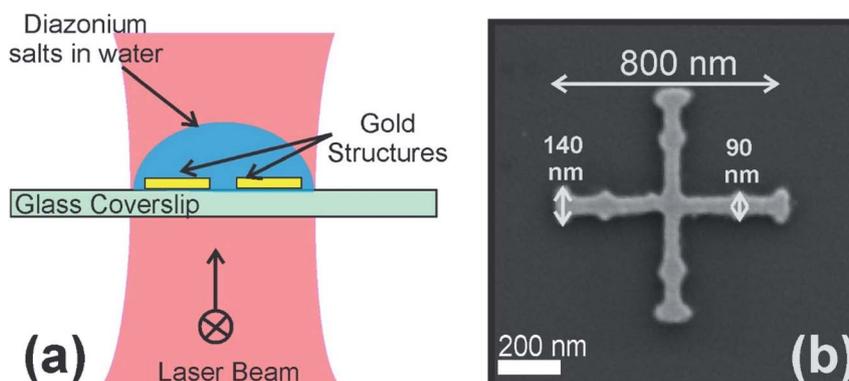


**Fig. 1** Distribution of the near-field at 530 nm (a), 630 nm (b) and 800 nm (c) for a horizontal polarization. The corresponding isowavelength map is shown in (d) and represents the spatial distribution of the field for wavelengths varying from 500 nm to 800 nm with a crosshair structure outlined in black.

Two distinct optical setups were used to investigate the three LSPRs. An 800 nm femtosecond-pulsed laser is used to irradiate a single structure at a time without burning or destroying them. This setup also uses a piezoelectric nanopositioning stage combined with a  $100\times$  objective (N.A. = 0.95). This combination allows the  $1\ \mu\text{m}$  laser spot to be precisely positioned onto a single structure at a time. The CW lasers at 532 nm and 632.8 nm were coupled to a distinct setup that uses a  $20\times$  objective (N.A. = 0.5) that is slightly defocused producing a spot size of approximately  $8\ \mu\text{m}$  and irradiating multiple structures at the same time. While this setup has the advantage of irradiating many structures at once, the Gaussian nature of the beam produces varying results from the center of the spot towards the edge, which will be discussed further. A typical irradiation experiment is shown in Fig. 2a together with a single nanostructure produced by electron-beam lithography (Fig. 2b).

Using electron beam lithography, arrays of structures were produced in  $(50 \times 50)\ \mu\text{m}^2$  patches, as shown in Fig. 3a. The individual crosshair major axes measure 800 nm in length with 90 nm and 140 nm minor notches oriented perpendicularly to the longer branches (Fig. 2b). Three broad plasmon resonances can be identified in the far-field FDTD modelling at central wavelengths of 530 nm, 630 nm, and 800 nm (Fig. 3b). The experimental extinction spectrum only shows 2 resonances including a broader one centered at 610 nm that shows a shoulder around 550 nm.

The crosshair structures were first irradiated at 800 nm for 60 seconds at a power of 10 mW. The material grafted is composed of a thin film formed from 4-nitrobenzenediazonium tetrafluoroborate,<sup>38</sup> through either a cationic or a radical plasmon-induced polymerization pathway.<sup>39</sup> Briefly, in the cationic process, heating under room temperature conditions



**Fig. 2** (a) Depiction of a grafting experiment. (b) Produced individual structures.



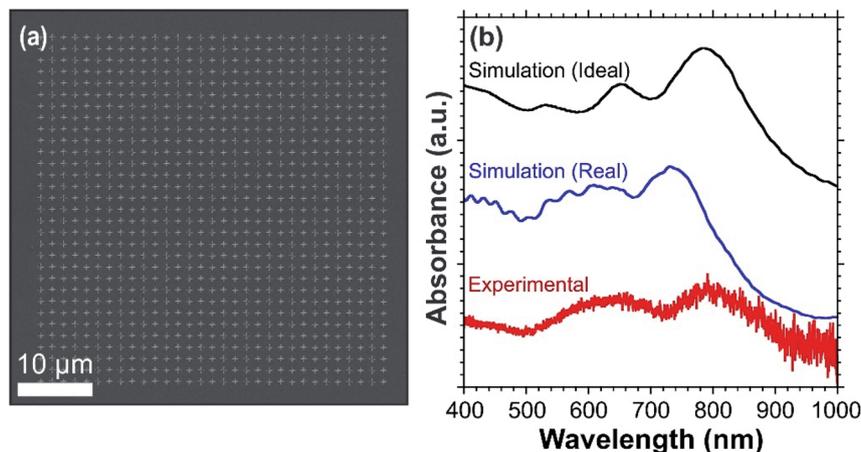


Fig. 3 (a) Representative SEM image of an array of dendrimers used for grafting with a single structure shown in the inset. (b) FDTD far-field scattering for ideal (defect-free) and real structures (as shown in Fig. 1b) and experimental absorption measurements for a linearly polarized light oriented horizontally or vertically with respect to the structure axes.

or light-induced decomposition of the solution-containing salt induces the formation of cations by heterolytic dediazonium that react with gold surfaces. Other aryl cations present in solution further react with the first grafted groups forming a polyaryl layer on the surface.<sup>40</sup> This cation-induced reaction is generally referred to as a spontaneous grafting and may lead to the formation of a monolayer at the surface of the substrate.<sup>39,41</sup> In the radical process, plasmon-induced electron transfer from

the gold structures to the diazonium cation yields the formation of aryl radicals that in turn react with the surface to form an aryl molecular layer. Other radicals react with the first layer to form a thin polymer film whose thickness varies with the irradiation conditions and local field enhancements. Both mechanisms presumably yield the formation of polymer thin films in the area where the field is maximum. However, the local heating effect from a cationic process will dissipate very quickly in the solution and because of the good thermal conductivity of the gold structures, the grafting will be more homogeneous over the surface of the structure. Additionally, the light decomposition of the salts occurs predominantly in the ultra-violet range and should be limited upon exposure at 800 nm. The grafting at 800 nm should essentially be driven by a radical mechanism. Fig. 4a and b show the SEM images after irradiation at  $\lambda = 800$  nm with a horizontal and vertical polarization, respectively. Polarization dependence is the most apparent here, where the orthogonal irradiations produced similar grafting patterns rotated by  $90^\circ$ . These experimental results can be compared to the FDTD calculations shown in Fig. 1c and 4c for ideal and real structures, respectively. The plasmon-induced grafting of the diazonium salts is predominantly localized around the two minor notches on the perpendicular branch with respect to the input polarization. While the irradiance is higher on the 800 nm setup (by a factor of 100–1000), the femtosecond pulses allow the structures to be irradiated without burning or melting them. The same localized molecular patterning can be performed over an array of structures irradiated under similar conditions, as shown in Fig. S1,† highlighting the reproducibility and homogeneity of the plasmon-induced reaction.

Furthermore, as shown in the inset of Fig. S1† with a reference structure immersed with the same solution but without any laser irradiation, this experiment indicates that there is no evidence of spontaneous grafting at the surface of the cross-hairs and that the grafting is only plasmon-driven, as expected.

Similar structures were irradiated with a continuous-wave laser at 632.8 nm with a spot size of approximately 8  $\mu\text{m}$  in

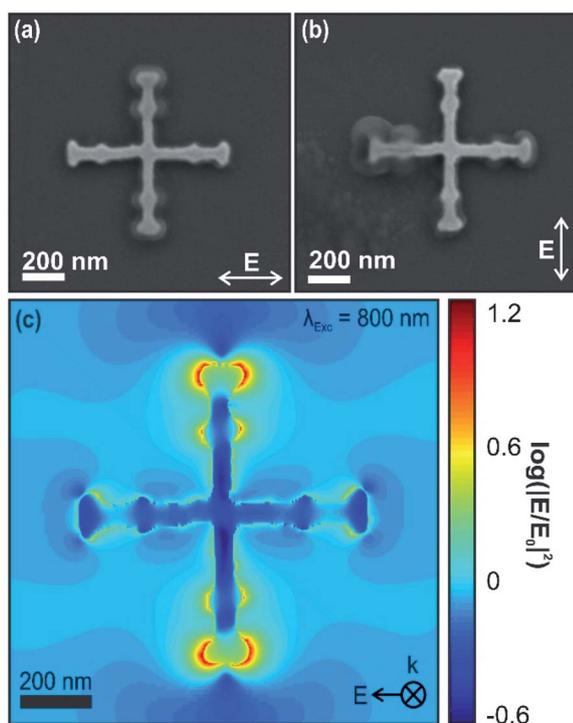


Fig. 4 SEM images of the dendrimers after irradiation with an 800 nm fs-pulsed laser, for both (a) horizontally and (b) vertically polarized light. (c) FDTD modelling of the near-field distribution upon irradiation at 800 nm for the real structure with a horizontal polarization.



diameter. A manual control stage was used, allowing for the irradiation of multiple structures at a time. This led to some variety in the grafting development based on the energy dose that each structure received relative to its position in the laser spot.

Fig. 5a and b show the polarization dependent grafting spatial distributions for both horizontally (Fig. 5a) and vertically (Fig. 5b) polarized fields. The grafted material has been colorized for the corresponding polarizations in Fig. 5c and d. Remarkably, the grafting occurs on both axes. However, the major axis orthogonal to the input polarization shows the thicker grafting. This is again in agreement with the FDTD calculations in Fig. 1b for the ideal and real structures and Fig. 5e for the real structure where both major axes show areas of large confinement yet with larger confinement on the axes orthogonal to the input polarization. In order to estimate the effect of the irradiation power over grafting, functionalized

areas were imaged at the center of the beam and in the outer edge of the Gaussian beam.

At the center of the 632.8 nm beam (Fig. S2a, b and S3a, b†) grafting is homogeneous over the whole structure and it is difficult to differentiate the areas with different field enhancements. In contrast, Fig. S2a, c and S3a, c† correspond to the area at the edge of the 632.8 nm Gaussian beam; the functionalization is partial and shows grafting along the orthogonal axis up to the first notch and around the larger notch of the axis parallel to the input polarization, in good agreement with Fig. 1b. This indicates that the threshold for plasmon-mediated grafting is relatively low. By fine tuning the energy density at the surface, the amount of material grafted can be controlled. This can be achieved by either reducing the irradiation time or the irradiance.

Finally, the last experiment was performed with an irradiation at 532 nm, a wavelength for which FDTD predicts a third

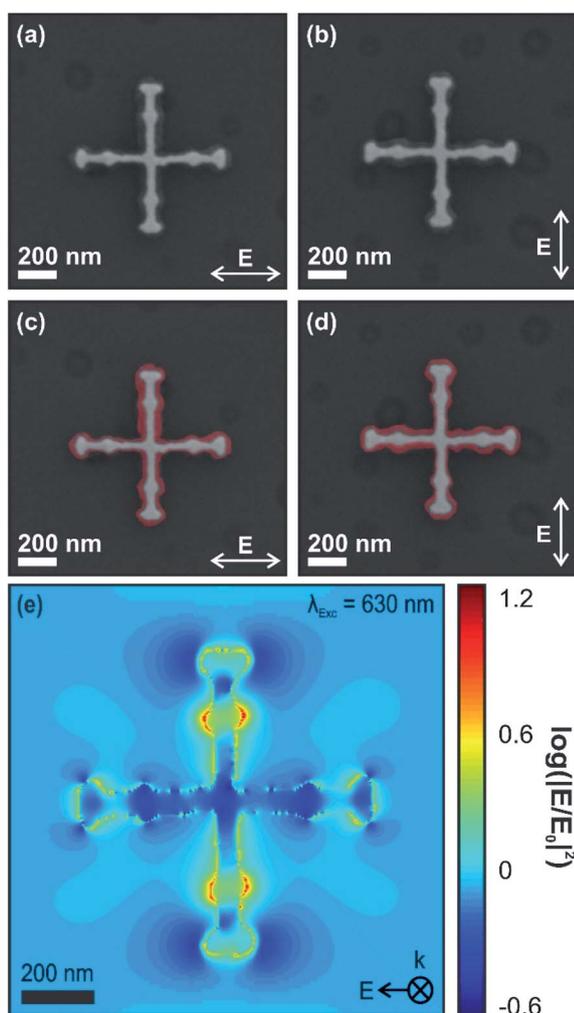


Fig. 5 SEM images of the structures after irradiation using a 632.8 nm continuous-wave laser, for both (a) horizontally and (b) vertically polarized light. (c and d) The same images but colorized in red to highlight the areas on which grafting mainly occurred. (e) FDTD modelling of the near-field distribution upon irradiation at 630 nm for the real structure with a horizontal polarization.

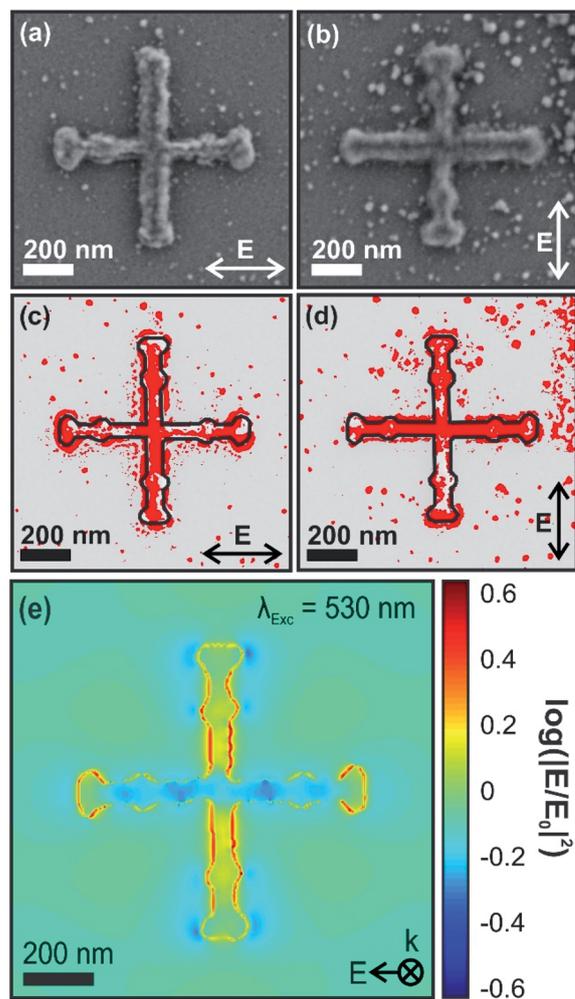


Fig. 6 SEM images of the crosshairs after irradiation using a 532 nm laser both (a) horizontally and (b) vertically polarized. (c) and (d) the colorized images of (a) and (b), respectively, with the outline of the cross in black as a guide for the eye. (e) FDTD modelling of the near-field distribution upon irradiation at 530 nm for the real structure with a horizontal polarization. The reference background image used to generate (c) and (d) is provided in Fig. S4.†



but weaker resonance. The extinction spectra (Fig. 2b) show that the resonance appears to be superimposed with a broader resonance centered at 630 nm.

Fig. 6a and b show the SEM images of individual nanostructures after irradiation in the presence of the diazonium solution with a horizontal and a vertical polarization, respectively. Fig. 6c and d show treated images obtained by subtracting the bare non-functionalized structure (Fig. S4†) and that show the spatial distribution of the polyaryl films. Both major arms appear to experience grafting yet with predominant grafting along the major arms that are orthogonal to the input polarization. In both cases the gold structure is entirely covered by the polyaryl layers derived from the diazonium salt which appear thicker than when irradiated at 800 or 632.8 nm. Interestingly, the irradiation at 532 nm appears to induce polymerization on the substrate as well, as shown by the numerous aggregates located around the structures. The presence of such aggregates is not observed in any of the experiments performed at 632 nm and 800 nm. Presumably the presence of these aggregates over the gold structure and the glass surface is facilitated by spontaneous grafting enabled by the higher energy 532 nm photons that would yield more thermally induced decomposition and subsequent grafting through the cationic process.

The LSPR present at 532 nm originates from the width of the structures themselves (Fig. 1a and 6e), which is perpendicular to the irradiation polarization. This can be seen by the growth occurring on all the outer features, and predominantly on the entire arm, perpendicular to the irradiation polarization as highlighted in red in Fig. 6a–d where the notches cannot be distinguished anymore since they are completely covered by the polyaryl layer.

The polarization dependent grafting demonstrates the effectiveness of the nanostructure's response upon irradiation. Fig. S5,† shows a series of crosshair structures located at the edge of the irradiation beam polarized vertically (Fig. S5a†) and horizontally (Fig. S5b†) with intensity increasing from left to right.

The bare structures on the left-most of Fig. S5a and b† are out of the beam while they transition to irradiation with the highest power on the right. These experiments show again that the major axes oriented perpendicular to the polarization of the impinging beam experience the most functionalization where the grafted molecules completely cover the major axis masking even the two smaller orthogonal notches.

The progression of grafting across the surface indicates that the density of hot electrons generated upon irradiation at 532 nm is proportional to the energy dose. It can also be seen in these images that the structures on the left have no molecules adhered to either the surface of the structures, or the glass substrate itself. On the right, however, there is a large grafting visible on the structures as well as organic aggregates adsorbed on the surface. This demonstrates that all the grafting is occurring due to the irradiation, and not spontaneously. While the 532 nm grafting had the lowest irradiance of the three tests at  $1.79 \times 10^3 \text{ W cm}^{-2}$ , grafting was more prominent for equivalent irradiation times (60 s). This is caused by the LSPR, which

covers a larger surface area of the crosshair structure than the other two resonances, in synergy with the cationic process as evidenced by the numerous organic deposits observed at the surface of the glass coverslip closer to the center of the laser spot.

In summary, the different grafting patterns based on the irradiation wavelength are shown. The electromagnetic enhancement hotspots from the plasmon resonances play an important role when comparing these different grafting patterns. By changing the irradiation source as well as polarization, molecules were grafted to specific areas of the surface. Although this work considers only a selected diazonium salt, there is potential to use these structures to graft up to six different molecules onto the same surface using three wavelengths and two polarization states. This versatility can be taken advantage of for molecular patterning. This opens new avenues in molecular electronics and biosensing where distinct grafted molecules within a nanoscale structure can display different functions.

## Conclusion

Crosshair nanostructures with three plasmon resonances were designed and fabricated in order to facilitate plasmon-mediated reactions in the visible wavelength region. The structures were successfully employed for the surface specific grafting of a diazonium salt to the surface using wavelengths of 532 nm, 633 nm, and 800 nm. For each of the wavelengths, distinct grafting patterns can be observed around the structures. The polarization dependence of grafting was also tested for these three wavelengths to which the structures responded according to incident polarization. These nanostructures offer a platform for a multitude of applications in both chemistry and nanoscale surface patterning due to the possibility of spatially controlling the functionalization of distinct molecules with a selected set of wavelengths and polarizations. Based on this approach, optical sensing devices with multiplexed specificity or bioconjugation could be developed.

## Author contributions

D. A. B. T. made the structures, collected the spectra, and performed the grafting experiments and SEM imaging. D. M. M performed the FDTD calculations. C. M., N. F. and F. L. L. directed the work conceptually and experimentally. All authors have participated in the writing of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors would like to gratefully acknowledge the Nanofabrication Facility at the University of Western Ontario for their assistance in fabricating the crosshair arrays. The authors would also like to thank the Natural Sciences and Engineering



Research Council (NSERC) for funding this project through grant (RGPIN-2020-06676).

## References

- 1 T. T. X. Ong, E. W. Blanch and O. A. H. Jones, *Sci. Total Environ.*, 2020, **720**, 137601.
- 2 D. Kurouski, A. Dazzi, R. Zenobi and A. Centrone, *Chem. Soc. Rev.*, 2020, **49**, 3315–3347.
- 3 G. Q. Wallace and F. Lagugné-Labarthe, *Analyst*, 2019, **144**, 13–30.
- 4 F. Zen, V. D. Karanikolas, J. A. Behan, J. Andersson, *et al.*, *Langmuir*, 2017, **33**, 4198–4206.
- 5 J. Zhou, T. Yang, J. Chen, C. Wang, *et al.*, *Coord. Chem. Rev.*, 2020, **410**, 213218.
- 6 H. Zhang, S. Duan, P. M. Radjenovic, Z.-Q. Tian, *et al.*, *Acc. Chem. Res.*, 2020, **53**, 729–739.
- 7 M. Sharifi, S. H. Hosseinali, R. Hossein Alizadeh, A. Hasan, *et al.*, *Talanta*, 2020, **212**, 120782.
- 8 G. Q. Wallace, S. T. Read, D. M. McRae, S. M. Rosendahl, *et al.*, *Adv. Opt. Mater.*, 2018, **6**, 1701336.
- 9 G. Q. Wallace, H. C. Foy, S. M. Rosendahl and F. Lagugné-Labarthe, *J. Phys. Chem. C*, 2017, **121**, 9497–9507.
- 10 N. Bareza Jr, K. K. Gopalan, R. Alani, B. Paulillo, *et al.*, *ACS Photonics*, 2020, **7**, 879–884.
- 11 J.-F. Li, C.-Y. Li and R. F. Aroca, *Chem. Soc. Rev.*, 2017, **46**, 3962–3979.
- 12 M. Bauch, K. Toma, M. Toma, Q. Zhang, *et al.*, *Plasmonics*, 2014, **9**, 781–799.
- 13 T. Ming, L. Zhao, Z. Yang, H. Chen, *et al.*, *Nano Lett.*, 2009, **9**, 3896–3903.
- 14 M. Tavakkoli Yarak, S. Daqiqeh Rezaei and Y. N. Tan, *Phys. Chem. Chem. Phys.*, 2020, **22**, 5673–5687.
- 15 D. A. B. Therien, R. Hou and F. Lagugné-Labarthe, *Plasmonics*, 2020, **15**, 507–515.
- 16 D. M. McRae, D. A. B. Therien, R. Hou, G. Q. Wallace, *et al.*, *ACS Appl. Nano Mater.*, 2020, **3**, 3922–3929.
- 17 X. Lin, Y. Liu, K. Wang, C. Wei, *et al.*, *ACS Nano*, 2018, **12**, 689–696.
- 18 J. Butet, P.-F. Brevet and O. J. F. Martin, *ACS Nano*, 2015, **9**, 10545–10562.
- 19 S. Yu and P. K. Jain, *Angew. Chem., Int. Ed.*, 2020, **59**, 2085–2088.
- 20 M. Ha, J.-H. Kim, M. You, Q. Li, *et al.*, *Chem. Rev.*, 2019, **119**, 12208–12278.
- 21 I. C. Bicket, E. P. Bellido, D. M. McRae, F. Lagugné-Labarthe, *et al.*, *ACS Photonics*, 2019, **6**, 2974–2984.
- 22 D. M. McRae, K. Jeon and F. Lagugné-Labarthe, *ACS Omega*, 2018, **3**, 7269–7277.
- 23 S. Pal, A. Dutta, M. Paul and A. Chattopadhyay, *J. Phys. Chem. C*, 2020, **124**, 3204–3210.
- 24 D. Hetemi, V. Noel and J. Pinson, *Biosensors*, 2020, **10**, 4.
- 25 S. Wallentine, S. Bandaranayake, S. Biswas and L. R. Baker, *J. Phys. Chem. A*, 2020, **124**, 8057–8064.
- 26 C. Lu, J. Li, J. Yan, B. Li, *et al.*, *Appl. Mater. Today*, 2020, **20**, 100744.
- 27 S. Yu and P. K. Jain, *Nat. Commun.*, 2019, **10**, 2022.
- 28 Y. Sang, B. Liu, T. Tao, D. Jiang, *et al.*, *Semicond. Sci. Technol.*, 2020, **35**, 025017.
- 29 Y.-S. Huang, Y.-C. Hsiao, S.-H. Tzeng, Y.-H. Wu, *et al.*, *Nano Energy*, 2020, **77**, 105267.
- 30 D. Ghosh, K. Roy, K. Sarkar, P. Devi, *et al.*, *ACS Appl. Mater. Interfaces*, 2020, **12**, 28792–28800.
- 31 D. Kumar, C. H. Park and C. S. Kim, *J. Mater. Chem. A*, 2020, **8**, 5734–5743.
- 32 I. Tjunelyte, I. Kherbouche, S. Gam-Derouich, M. Nguyen, *et al.*, *Nanoscale Horiz.*, 2018, **3**, 53–57.
- 33 X. Zhou, J. Wenger, F. N. Viscomi, L. Le Cunff, *et al.*, *Nano Lett.*, 2015, **15**, 7458–7466.
- 34 H. Gehan, L. Fillaud, M. M. Chehimi, J. Aubard, *et al.*, *ACS Nano*, 2010, **4**, 6491–6500.
- 35 Y. Wang, S. Wang, S. Zhang, O. A. Scherman, *et al.*, *Nano Res.*, 2018, **11**, 6384–6390.
- 36 D. M. McRae and F. Lagugné-Labarthe, *Nat. Nanotechnol.*, 2019, **14**, 922–923.
- 37 E. Palik, *Handbook of Optical Constants of Solids*, Academic Press, Burlington, 1997.
- 38 J. Pinson and F. Podvorica, *Chem. Soc. Rev.*, 2005, **34**, 429–439.
- 39 A. Mesnage, X. Lefèvre, P. Jégou, G. Deniau, *et al.*, *Langmuir*, 2012, **28**, 11767–11778.
- 40 M. Nguyen, A. Lamouri, C. Salameh, G. Levi, *et al.*, *Nanoscale*, 2016, **8**, 8633–8640.
- 41 D. R. Jayasundara, R. J. Cullen, L. Soldi and P. E. Colavita, *Langmuir*, 2011, **27**, 13029–13036.

