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Composition and properties of RF-sputter deposited titanium dioxide thin films†

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The photocatalytic properties of titania (TiO₂) have prompted research utilising its useful ability to convert solar energy into electron–hole pairs to drive novel chemistry. The aim of the present work is to examine the properties required for a synthetic method capable of producing thin TiO₂ films, with well defined, easily modifiable characteristics. Presented here is a method of synthesis of TiO₂ nanoparticulate thin films generated using RF plasma capable of homogenous depositions with known elemental composition and modifiable properties at a far lower cost than single-crystal TiO₂. Multiple depositions regimes were examined for their effect on overall chemical composition and to minimise the unwanted contaminant, carbon, from the final film. The resulting TiO₂ films can be easily modified through heating to further induce defects and change the electronic structure, crystallinity, surface morphology and roughness of the deposited thin film.

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Introduction

Titania (TiO₂) has been a major area of research since the discovery of its photocatalytic properties in 1972.¹ Since then, TiO₂ has been one of the most widely studied semiconductors, driven largely due to its non-toxic nature, low cost, stability, and varied methods of synthesis and property control.^{2–4} The broad range of possible applications stems from its versatile photocatalytic properties. The activity of this material under UV light has led to its inclusion in many sun protection applications,⁵ sensors,² integrated optical devices,⁶ photo-oxidation of pollutants and dangerous chemicals,⁷ and photocatalytic processes for energy storage.⁸ All of these applications utilise TiO₂ in the form of nanoparticulate coatings,⁹ making scalable, affordable synthesis of such coatings valuable, especially if they can be modified for specific properties.

The synthesis of TiO₂ amorphous films, thin layers of nanoparticle sized TiO₂ has been achieved through a variety of techniques such as chemical vapour deposition (CVD),¹⁰ sol–gel processing^{11,12} atomic layer deposition¹³ and direct current (DC) and radio frequency (RF) magnetron sputtering processes.^{14,15} These final two sputtering methods have been widely explored due to their stability, reproducibility, ease of use and deposition control over a variety of substrates. The films deposited through this method are commonly dense, closed layers of

nanoparticulate TiO₂. Nanoparticle Titania has shown particular promise across the range of possible applications and also offers a level of control over characteristics such as particle size,¹⁶ phase-type^{17,18} and stoichiometry.¹⁷

The use of well-understood TiO₂ nanoparticle thin films is especially useful for surface science studies utilising photoelectron spectroscopy. Several valuable techniques for probing electronic structure such as X-ray photoelectron spectroscopy (XPS) and ultraviolet photoelectron spectroscopy (UPS) require precision knowledge of a sample's elemental composition and electronic structure to yield meaningful results.^{19,20} Many photocatalytic systems require a robust substrate–catalyst electronic interaction to prevent changes in the performance of the catalyst due to agglomeration; one method proposed to achieve this is defect rich surfaces.^{21,22} Ti³⁺ defects have been explored as promising binding sites for heterogeneous catalysts that offer strong electronic interaction to prevent agglomeration.^{23,24}

The topography, crystallinity and particulate characteristics have been well established in sputtered thin films using atomic force microscopy (AFM),^{6,25} scanning electron microscopy (SEM)^{18,26} and X-ray diffraction (XRD).^{6,16,18} However, the elemental composition of the final coating is relatively unexplored for RF sputtered TiO₂. Carbon-containing substances like hydrocarbons are considered as a contaminant for many methods of electron spectroscopy and is present in considerable quantities for many commonly used methods of TiO₂ synthesis. Hence, an RF sputtered deposition procedure resulting in low carbon percentage is desirable. Single crystal TiO₂ is also capable of addressing the issue of contamination but is prohibitively expensive for applications on a larger scale.

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TiO₂ can be either amorphous or has one of three crystalline forms: anatase, brookite or rutile. Layers or films of titania can be either pure or mixtures of these forms with synthetic methods often influencing the structure of the coating through control of factors such as deposition temperature,²⁶ or by post-deposition treatments.^{6,27} The surface morphology and crystal phase composition of TiO₂ films are also modifiable through post-deposition calcination and have been shown to influence and improve the photocatalytic properties of deposited films.²⁸

The present study demonstrates a method of depositing RF sputtered TiO₂ thin films onto Si(110) substrates, with a range of target pre-treatments, deposition lengths and annealing as post-treatment. Systematic studies into elemental composition, morphology, crystallinity and chemical state of the nanoparticulate TiO₂ were conducted to determine their composition, contaminants and ease of surface modification with Ti³⁺ defects. The results indicate that one method, in particular, produced a TiO₂ nanoparticulate film capable of low-cost scalability, minimal carbon contamination in the final film and simple modification of surface defects, crystallinity and surface morphology.

Experimental details

Preparation of TiO₂ thin films

TiO₂ thin films used in this work were prepared at room temperature using an RF magnetron sputtering system. High purity 2-inch (99.9%) TiO₂ targets were used to deposit TiO₂ nanoparticles onto 4-inch diameter Si(110) substrates. 2 TiO₂ targets were used, one having been used for several months, the other being brand new, to determine how target ageing affects the resultant deposition quality. The substrates were cleaned with ethanol and acetone for 15 min in an ultrasonic bath. The substrates were then dried under nitrogen. The deposition was done three times, with differing chamber/target conditions. Firstly after 1.5 hours of pre-sputtering to clean the target and chamber (designated 90m-RF), secondly after 30 min of pre-sputtering for cleaning (designated 30m-RF) and finally with an unused TiO₂ target after 30 min pre-sputtering (designated FT-RF). The sputter deposition was performed in all cases with a chamber pressure of at least 2×10^{-5} mbar, with 5 ccm pure argon flowing, 500 W magnetron power and a working distance of 14 cm. These conditions provided a deposition rate of approximately 180 nm h⁻¹. The deposition took place at room temperature with no substrate heating.

All samples were cut into 1 × 1 cm squares using a diamond-tipped silicon cutter to minimise surface contact and contamination. Particulate dust from cutting was removed with pressurised dry nitrogen. Samples were then mounted to molybdenum sample holders and inserted into the ultra-high vacuum chamber ($9.2\text{--}6 \times 10^{-10}$ mbar). The samples were characterised in this unaltered condition (designated by the As-made tag next to sample name) before the samples were heated through filament emission heating to 500 °C and held there for 10 minutes in an attempt to reduce film contaminants. XPS scans marked “heated” were taken almost immediately after heating.

Table 1 Sample series for comparison of deposition methods (measured As-made and heated 500 °C in vacuum) and vacuum heated series using an FT-RF sample

Deposition methods	FT-RF TiO ₂ – vacuum heated series, treatment
FT-RF	300 °C
30m-RF	500 °C
90m-RF	700 °C

Samples from all deposition methods were measured in As-made condition as well as after heating to 500 °C in UHV. Further, post-deposition heating was performed on a single sample prepared from the FT-RF method in vacuum at 300 °C, 500 °C and 700 °C (see Table 1) in succession with photoelectron measurements taken after 30 minutes of heating at a given temperature before increasing the temperature. The sample designated “300 °C and Ar sputtered” was heated to 300° as described previously and then immediately exposed to an argon ion sputter beam, exposing the sample surface with a dose of 6×10^{14} ions, controlled through exposure time dependant on ion current (see ESI S1†). FT-RF samples were also heated up to 1100 °C in atmospheric conditions for crystal phase structure investigations.

Additionally, electron spectroscopy measurements were taken from rutile (110) single crystal (SC). The SC TiO₂ was pre-treated at 877 °C (1150 K) *ex situ* to induce bulk defects, making the sample conductive and useful for electron spectroscopy. The SC TiO₂ again was heated to 625 °C *in situ* immediately before photoelectron measurements to produce surface defects, the XPS relative elemental composition of SC TiO₂ can be seen in Table S3.†

Spectroscopy

X-ray photoelectron spectroscopy (XPS) & ultraviolet photoelectron spectroscopy (UPS). Chemical composition of the prepared samples was undertaken using X-ray photoelectron spectroscopy (XPS). These measurements were taken with an X-ray source producing 1253.6 eV (Mg K α radiation) and 200 W power using a SPECS PHOIBOS-HSA300 hemispherical analyser (Berlin, Germany). Survey scans were taken at 40 eV pass energy and high-resolution scans were taken at 10 eV pass energy. The main adventitious carbon peak was used as an energy reference for all other species and set to 285 eV. All XP spectra were analysed with Casa XPS using a Shirley background and a hybrid Gaussian (30%) and Lorentzian (70%) function to fit all peaks in a method described previously.²⁹ Elemental sensitivity factors for an X-ray source at 54.7° were used is the analysis process.³⁰ Sample heating was performed either under Ultra-High Vacuum (UHV) conditions or in an air-atmosphere furnace, as specified in the results. UPS measurements were performed in the same UHV chamber and with the same SPECS hemispherical analyser. The probe source used is a two-stage cold cathode gas discharge source from MFS (Clausthal-Zellerfeld, Germany) capable of generating He I line UV photons and metastable He atoms.



Table 2 XPS relative elemental compositions of TiO₂ samples prepared with 90 min pre-sputtering (90m-RF), 30 min pre-sputtering (30m-RF) and 30 min pre-sputtering and a fresh TiO₂ target (FT-RF). Samples designated As-made has received no further treatment after deposition. Samples designated – heated, have been heated to 500 °C in UHV conditions before measurement

XPS relative elemental composition (%)						
	Carbon	Oxygen	Titanium	Silicon	Fluorine	Sodium
FT-RF-As-made	20.2 ± 0.1	60.4 ± 0.1	18.2 ± 0.1	0.2 ± 0.1	0.9 ± 0.1	0.00
30m-RF-As-made	14.6 ± 0.1	61.3 ± 0.1	18.5 ± 0.1	0.1 ± 0.1	0.00	5.5 ± 0.1
90m-RF-As-made	26.8 ± 0.1	54.8 ± 0.1	16.7 ± 0.1	1.1 ± 0.1	0.00	1.7 ± 0.1
FT-RF – heated	5.9 ± 0.1	72.1 ± 0.1	20.1 ± 0.1	0.5 ± 0.1	0.00	1.5 ± 0.1
30m-RF – heated	8.7 ± 0.1	57.1 ± 0.1	20.8 ± 0.1	5.0 ± 0.1	0.00	9.2 ± 0.1
90m-RF – heated	20.5 ± 0.1	57.4 ± 0.1	19.1 ± 0.1	0.6 ± 0.1	0.00	2.3 ± 0.1

X-ray diffraction (XRD). XRD was performed on a range of samples in as-made and heated (both in-vacuum and atmospheric) condition to determine the crystallinity of the deposited TiO₂ nanoparticle films. These measurements were performed using a Bruker D8 Endeavor instrument with a Cobalt X-ray source at 1.78896 nm. Results were compared to TiO₂ anatase and rutile spectra calculated using materialsproject.org crystal toolkit for TiO₂.^{31,32}

Microscopy

Once the levels of unwanted contaminants were determined, and the best candidate surface (FT-RF) was selected, further analysis of the morphological characteristics of the film was performed using AFM and SEM techniques. Samples were prepared by identical cutting and cleaning procedures, with selected samples being heated in an air-atmosphere furnace at the required temperature before mounting onto flat or vertical SEM stubs as necessary. All SEM measurements were performed on the samples as they are.

Atomic force microscopy (AFM). AFM scans were performed utilising a Multimode 8 AFM with Nanoscope V controller at scan sizes of 1 × 1 μm and 3 × 3 μm on both as-made and heated (500 °C) samples to investigate surface roughness and morphological characteristics. The analysis was performed using Bruker's Nanoscope Analysis package to extract surface roughness values to determine changes in the nanoparticle size and surface roughness after post-deposition heating. The

surface roughness was measured using R_q , the root-mean-square roughness and R_a , the mean height deviation across a line profile.

Scanning electron microscopy (SEM). SEM scans were performed using an FEI Inspect F50 instrument with 30 kV on 0.5 × 1 cm TiO₂ samples mounted onto either a vertical SEM mount to determine the deposited film thickness or horizontally-mounted for morphological investigation. Thickness measurements were taken at several sites along the sample edge and averaged.

Results and discussion

Chemical composition

XP spectra were taken at the C 1s, O 2s, Ti 2p, Si 2p, F 1s and Na 1s regions for all samples. The breakdown of peak positions found in each region is presented in detail in the ESI (S2†) and the relative intensities in Table 2. The carbon in the sample is mostly attributed to adventitious hydrocarbons, and the C 1s peak is assigned to 285 eV. This peak is very consistent in its energy position and therefore, suitable for use in calibration.³³ The oxygen signal is attributed mostly to titanium dioxide (530.3 ± 01 eV) with some minimal contribution from silicon dioxide (532.3 ± 0.1 eV), the high relative percentage of oxygen is discussed later in this text when describing the results shown in Fig. 1. The titanium signal is exclusively due to the nanoparticle thin film, and the dominant species (Ti⁴⁺) is assigned to

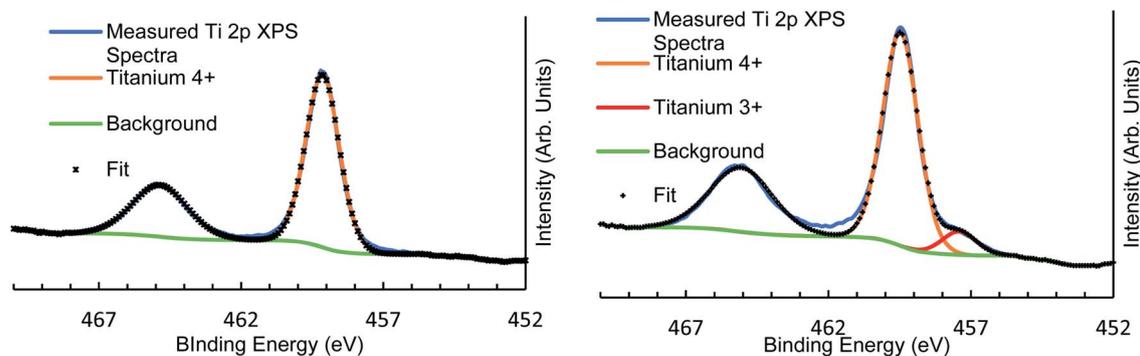


Fig. 1 XPS high-resolution scan fitting for vacuum heated Ti 2p region showing As-made (left) and 500 °C heated (right) with Ti³⁺ defect showing in the heated sample at around 457 eV binding energy.



Table 3 XPS Relative elemental compositions from an FT-RF sample, heated under UHV conditions to increasing temperatures to determine the change in C contribution through sequential heating. It can be seen that the level of C contribution in an As-made sample is significantly reduced by 300 °C, decrease through 500 °C and show a slight increase at 700 °C. Sputtering an FT-RF sample with Ar ions at a set dose after heating to 300 °C shows a reduction in C contribution over solely heating, as well as an increase in Ti³⁺ defects

FT-RF:vacuum heated series	Carbon	Oxygen	Titanium	Ti ³⁺ defect	Silicon	Fluorine
As-made	26.0 ± 0.1	54.7 ± 0.1	18.1 ± 0.1	0.00	0.5 ± 0.1	0.7 ± 0.1
300 °C	5.1 ± 0.1	72.1 ± 0.1	21.4 ± 0.1	1.2 ± 0.1	0.6 ± 0.1	0.7 ± 0.1
500 °C	4.8 ± 0.1	71.6 ± 0.1	22.8 ± 0.1	1.7 ± 0.1	0.8 ± 0.1	0.00
700 °C	5.6 ± 0.1	70.3 ± 0.1	22.5 ± 0.1	1.6 ± 0.1	1.6 ± 0.1	0.00
300 °C + Ar sputtered	4.4 ± 0.1	68.8 ± 0.1	26.1 ± 0.1	4.5 ± 0.1	0.8 ± 0.1	0.00

459.3 ± 0.1 eV with titanium defects induced through sample treatments (Ti³⁺) assigned to 457.6 ± 0.1 eV. The silicon signal is due to SiO₂ from the silicon wafer used as a substrate and is found at 102.4 ± 0.1 eV. Both sodium (1071.2 ± 0.1 eV) and fluorine (685.4 ± 0.1 eV) were contaminants of unknown origin with fluorine only found in one sample, while Sodium appeared in varying concentrations in all but FT-RF-As-made and is likely due to sample handling prior to the XPS measurements.

The XPS scans of these samples (Fig. 1, fits can be found in Table S2†) revealed several trends and characteristics. Across all samples, the ratio of Ti 2p orbital signature to O 1s is around 3.1 : 1 (O : Ti) with O accounting for an average of 60% of the measured signal while Ti accounts for approximately 20%. These values, while not stoichiometrically expected, are both consistent between the three prepared surfaces as well as comparisons to previously measured P-25 and atomic layer

deposited TiO₂ surfaces measured using the same experimental equipment³⁴ and as such are seen as representative of TiO₂. One possible reason for non-stoichiometric O concentration is adsorbed O species such as OH and H₂O. Another possible reason is an inaccuracy in the sensitivity factor applied for the O 1s XPS intensity.

Surface contaminants, a particular concern when using synthesised thin films for surface-sensitive measurements, were also determined through their characteristic XPS orbital signatures (Table 3). Carbon (1s) was the most apparent contaminant with every sample showing some noticeable percentage of carbon (between 6–26%) mainly due to adventitious carbon.³⁵ The highest levels of C 1s signal were found in as-made samples which had been exposed to the atmosphere after deposition and before undergoing heating, as would be expected. The 90m-RF sample was particularly carbonaceous in both as-made and post-heated form, indicating that a longer pre-cleaning time increases the carbon content within the final film, counter to the processes intent. The surface prepared through the fresh TiO₂ target (FT-RF) was the least carbonaceous of the three methods.

Silicon signal was below 1.1% for all samples; this elemental presence may be at least partly attributable to an intermixing of Ti and Si during the RF deposition. The only exception was for the 30m-RF-As-made, which showed approx. 5% Si after heating, probably due to film spallation³⁶ revealing some bare Si.

Table 4 Ti³⁺ defects as a proportion of total Ti 2p signal

Vacuum heated FT-RF	Ti ³⁺ defects (%)
300 °C	5.4%
500 °C	7.5%
700 °C	6.9%
300 °C + Ar sputtered	17.1%

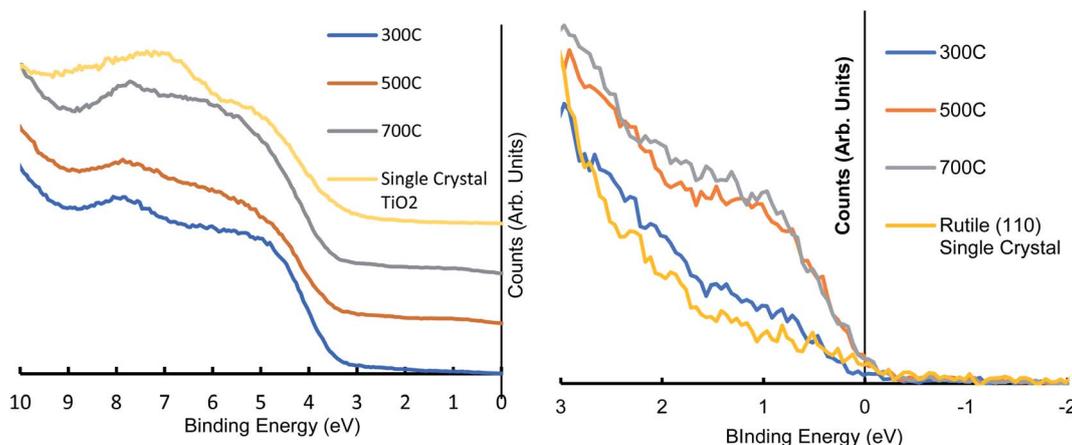


Fig. 2 UPS scans from a range of heated FT-RF samples and rutile (110) single crystal TiO₂ showing broad range electronic structure on the left (series are offset in the y-axis) and the near-Fermi valence band on the right.



This, coupled with the observation that while in previous studies on TiO₂ film depositions Si 2p were detected at a binding energy of 99.8 ± 0.3 eV,³⁴ all Si signal from these samples is observed at 102.3 ± 0.3 eV. This binding energy points towards the presence of organic silicon compounds. The RF-sputter deposition process likely results in the formation of small amounts of silicate species³⁷ or a Si–O–Ti compound formed during deposition.

Previous spectroscopic studies on systems utilising TiO₂ have had issues with Si and C signal interfering with sensitive data collection and analysis.³⁴ These contaminant species are found to be present in all samples prepared through RF sputter deposition. However, the sample prepared from the fresh target of TiO₂ showed negligible Si 2p signal and a very low concentration of C 1s, further lessened through heating in vacuum, making it the best candidate for sensitive surface spectroscopic studies.

Other contaminants were minimal. Fluorine (1s) was observed in the fresh target sample before heating at 1% of the surface signal. However, heating removed this signal to below detectable amounts. Sodium (1s) was also detected although the variability of Na in samples from the same deposition proves suggests this contamination is due to sample processing. However, after heating, all sodium-containing samples had

their sodium content increase, raising the possibility that some sub-surface Na may be drawn to the surface through heating processes. This interpretation would be consistent with the observed low surface energy of Na, causing it to migrate towards the surface of metal oxides during heating processes and electron beam exposure.^{38,39} Samples from the FT-RF:vacuum heated series were heated under UHV conditions to analyse any compositional/electronic changes that might be induced. Calcination of FT-RF samples was performed over a range of temperatures under UHV conditions (Table 3). It can be seen that calcination as low as 300 °C was able to produce a significant reduction in C XPS signal, with a further slight C reduction for heating at 700 °C. The increase in C at 700 °C was accompanied by an increased Si contribution, pointing toward the onset of film spallation at this temperature, consistent with previous reports on TiO₂ thin films.³⁶ This process exposes small regions of the Si substrate through cracks in the film, simultaneously exposing C impregnated in the bulk of the TiO₂ film. To further explore this, an FT-RF sample was heated to 300 °C before being exposed to a dose of sputtering Ar ions. Sputter treatment should preferentially remove any C present at the surface of the film, while bulk C signal is unaffected. The 300 °C + Ar sputter treatment as can be seen (Table 3) to result in the lowest C

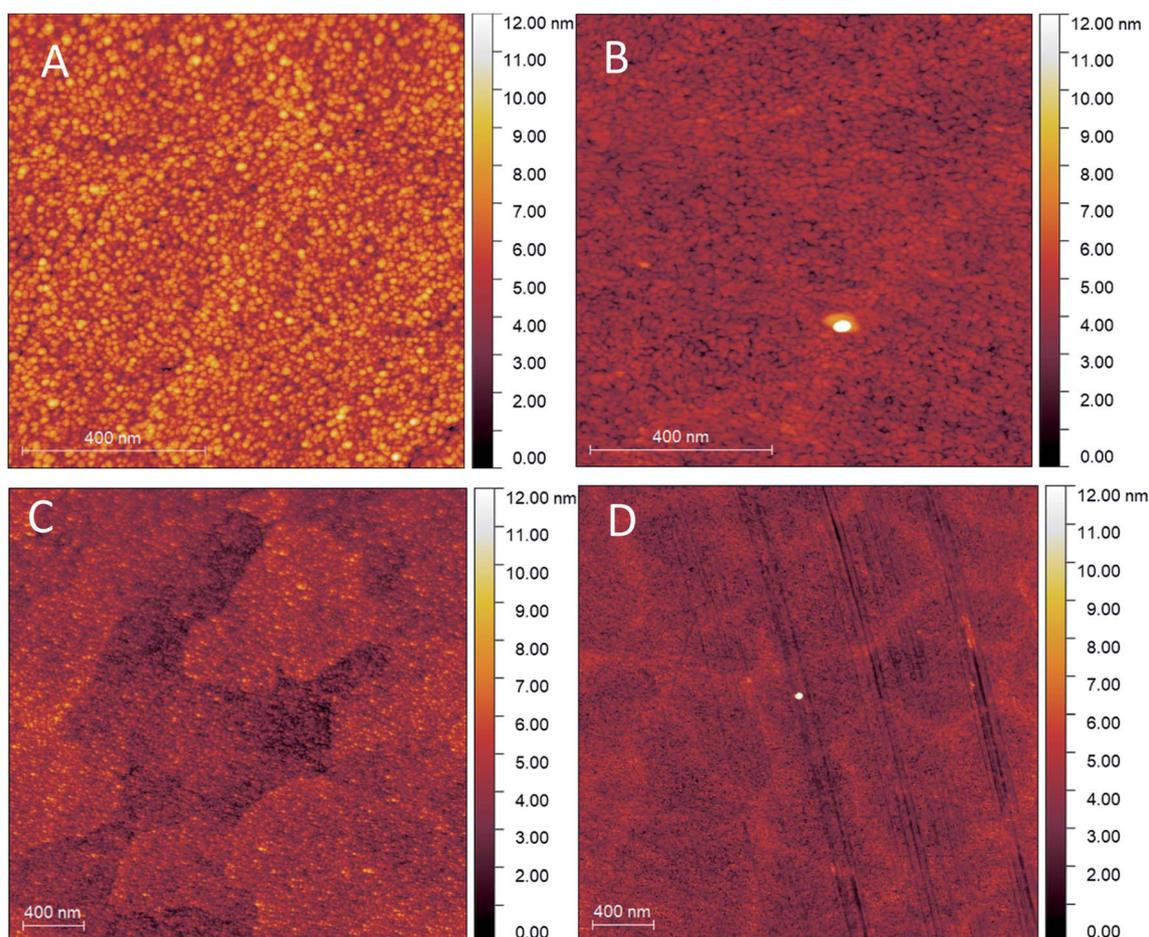


Fig. 3 AFM scans on (A) As-made, $1 \times 1 \mu\text{m}$ (B) 500 °C vacuum heated, $1 \times 1 \mu\text{m}$ (C) As-made, $3 \times 3 \mu\text{m}$ (D) 500 °C vacuum heated, $3 \times 3 \mu\text{m}$.



contribution compared to a purely heated sample at $4.4 \pm 0.1\%$ of XPS relative composition. The C signal in the $300\text{ }^\circ\text{C}$ + Ar sputter treatment is therefore assumed to be a baseline C contribution through RF sputtered deposition, as it would not be removable without compromising the integrity and coverage of the TiO_2 layer.

The XPS elemental relative concentrations seen in Table 3 show that heating under vacuum to higher temperatures induced Ti^{3+} defects at around $457.3 \pm 0.1\text{ eV}$ binding energy (Fig. 1). These defects increase as a relative proportion of total Ti 2p signal with heating, as seen in Table 4. The highest

proportion of Ti^{3+} defects are seen in the $300\text{ }^\circ\text{C}$ and Ar sputtered sample, which registered 17.1% of all Ti XPS signal as Ti^{3+} defect. The sputter treatment also decreased the overall O XPS signal in the sample, indicating that these defects were the result of oxygen vacancies induced through the sputter dose. Exposure to atmosphere quenched all previously measured Ti^{3+} signal. The relative ease with which such defects can be induced is a useful surface modification for use in studies of heterogeneous photocatalytic systems.

The UPS scans of the heated range reveal the changing electronic structure of the TiO_2 nanoparticles as they are heated, energies above 10 eV are affected by secondary electrons and as such are not considered here. The electronic features seen in heated RF- TiO_2 in the region from 3–9 eV are consistent with several previously reported studies examining the features found in DOS of modified TiO_2 and are reported to be from O(2p) species.^{40,41}

Krischok *et al.* identify the peak around 4.5 eV as being associated with the ionisation of oxygen states in the upper valence band, mainly from bridging oxygen in the TiO_2 , it can be seen that all four spectra in Fig. 2 share this peak's energy

Table 5 Average roughness characteristics from as-made and $500\text{ }^\circ\text{C}$ heated sputtered TiO_2 samples

Roughness parameters	$500\text{ }^\circ\text{C}$ vacuum heated (nm)	As-made (nm)
R_q	0.76 ± 0.01	1.06 ± 0.12
R_a	0.60 ± 0.03	0.84 ± 0.08

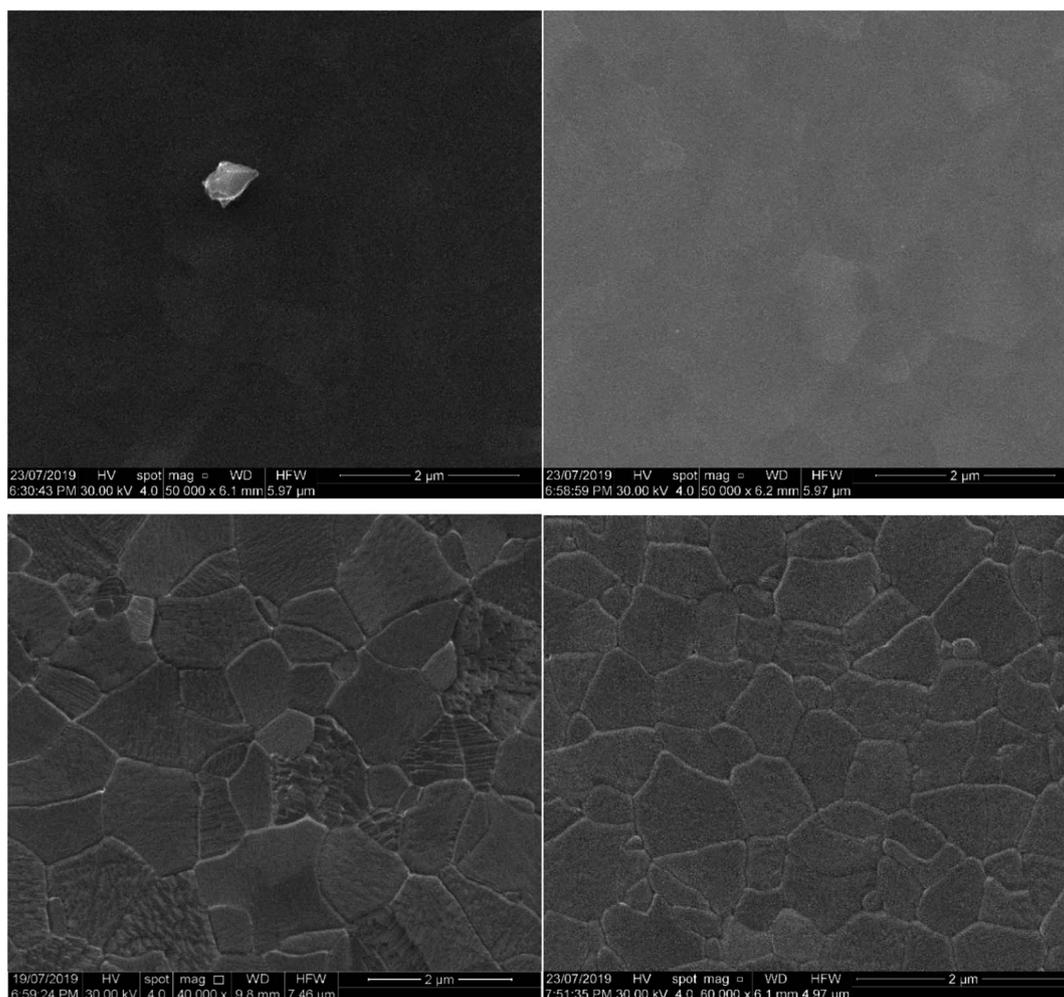


Fig. 4 SEM scans showing the morphological changes to FT-RF sputtered TiO_2 films as they are heated, clockwise from top left, $500\text{ }^\circ\text{C}$ 1 h, $600\text{ }^\circ\text{C}$ 1 h, $900\text{ }^\circ\text{C}$ 1 h, $1100\text{ }^\circ\text{C}$ 24 h. The particle shown in the $500\text{ }^\circ\text{C}$ image was included to aid focussing of the measurement.



position. The visible decrease in the 4.5 eV peak over the heated range of samples corresponds with a loss of the bridging oxygen, giving rise to the observed increase in Ti^{3+} defects in Table 4.

The 8 eV peak originates from oxygen species in the oxygen plane of the TiO_2 surface.^{42,43} While the single-crystal TiO_2 shares the 4.5 eV peak with RF- TiO_2 samples, the single crystal displays a shifted oxygen plane energy peak at around 7 eV compared to the RF- TiO_2 which appears at around 8 eV. This kind of variance in TiO_2 UPS spectra is not unexpected, as similar differences have previously been observed.^{44,45} One possible reason for such a shift could be attributed to the lower work function of the single crystal TiO_2 .⁴²

The near-Fermi region shows a clear peak centred at 1 eV is attributed to the Ti^{3+} defect and shows an apparent increase at higher temperatures for the RF-sputtered films, in agreement with XPS observations. The UPS spectra of the RF sputtered TiO_2 shows higher signal around 1 eV, attributed to Ti^{3+} defects, when compared to single-crystal TiO_2 showing its suitability for surface modification, even at lower calcination temperatures.

Morphological characterisations

Fig. 3 shows the uniformly distributed nature of the nanoparticulate film deposited through RF- TiO_2 deposition. SEM on a vertically mounted sample (Fig. S2†) showed the deposited TiO_2 film having a thickness of 150 ± 5 nm across the film's cross-section.

Comparison between Fig. 3A–D shows differences between the as-made and vacuum heated samples. The relatively smooth, close-packed nature of the film was observed across the sample surface, and in as-made condition consists of roughly spherical particles approximately 25–45 nm in diameter (Fig. 3A). While the sample heated to 500 °C appears to show a lack of spherical particles and the appearance of larger particles of arbitrary shape, possibly through the ripening of the TiO_2 under vacuum heating conditions. The 3 μm scans (Fig. 3C and D) show more significant regions separated by boundaries on the surface of increased height in both the as-made and vacuum heated samples. Again the vacuum heated sample displays less defined particles and a noticeable flattening of the film surface when compared to the as-made sample viewed at the same magnification.

The roughness parameters (R_q root-mean-squared average roughness, R_a mean surface roughness) values for the heated samples have also decreased from the as-made samples to the heated samples (see Table 5). This change confirms that the RF- TiO_2 deposited film has undergone some level of rearrangement from the as-made, nanoparticulate form into a flatter, more aligned surface. High-resolution images in the ESI (Fig. S3†) show that these spherical particles grow to larger sizes after heating to higher temperatures. Previous work has attributed these changes to an increase in crystallinity of TiO_2 and smaller particles' ripening into larger ones.^{6,16,46}

The large agglomerations of particles seen in Fig. 3C seem to have transitioned into the raised lines visible in Fig. 3D that divide the film into rough regions. These same phenomena can

be observed in Fig. 4; once calcination temperatures reached >900 °C the resultant condition of the coating was that of well-defined grain boundaries containing regions of varying size. At 1100 °C, some of these regions showed a visibly distinct texture with increased roughness, potentially related to a crystal phase transition observed in samples heated to this temperature.

Crystal structure

Samples measured with XRD are shown with offset intensities in Fig. 5 and spanned a range from (i) Si, (ii) as-made RF- TiO_2 , (iii) as-made atomic layer deposition TiO_2 , (iv) 500 °C atmosphere heated RF- TiO_2 (1 h), (v) 600 °C atmosphere heated RF- TiO_2 (1 h), (vi) 900 °C atmosphere heated RF- TiO_2 (3 h) and (vii) 1100 °C atmosphere heated RF- TiO_2 (24 h) and (viii) 1100 °C atmosphere heated RF- TiO_2 (24 h). The reference spectra for anatase and rutile TiO_2 calculated using the materials project are displayed in Fig. 5, showing the reference peaks for the respective TiO_2 phases.

The change in crystallinity in systems such as reactively sputtered TiO_2 films and sol-gel crystallised TiO_2 have previously found the phase transition to rutile phase to occur at around 600 °C with film thickness and temperature being critical factors in the phase transition.^{17,37} The system studied here shows markedly different crystallisation behaviour. We see in Fig. 5 that the onset of anatase phase TiO_2 appears from 500 °C heated samples, up to 900 °C with no corresponding rutile

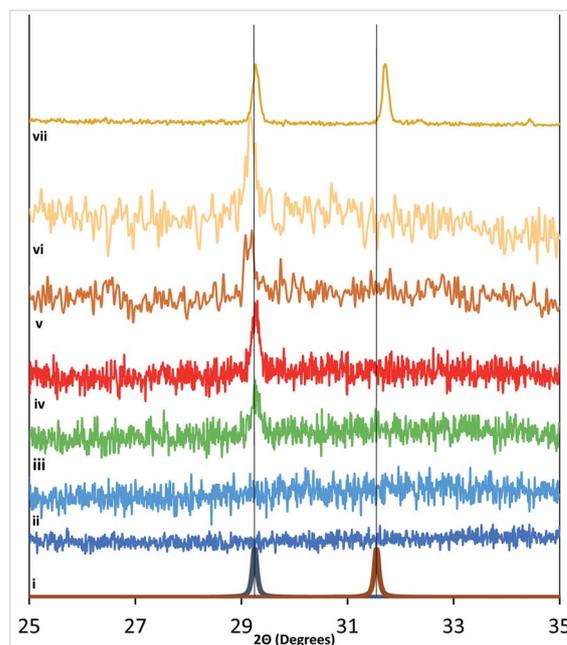


Fig. 5 Offset XRD scans on as-made and heated samples showing the transition from amorphous TiO_2 towards anatase phase. Rutile phase formation was detected only within the 1100 °C heated sample. Anatase and rutile main XRD peaks shown on the x-axis, in blue and orange respectively. XRD spectra shown are XRD spectra shown are (i) Si, (ii) as-made RF- TiO_2 , (iii) as-made atomic layer deposition TiO_2 , (iv) 500 °C atmosphere heated RF- TiO_2 (1 h), (v) 600 °C atmosphere heated RF- TiO_2 (1 h), (vi) 900 °C atmosphere heated RF- TiO_2 (3 h) and (vii) 1100 °C atmosphere heated RF- TiO_2 (24 h).



phase emerging. The emergence of Rutile crystal peaks in the XRD is seen only in 1100 °C samples heated for > 18 hours. This high temperature required produced samples with distinct peaks for anatase and rutile phases, with an approximate 1 : 1 ratio in the peak areas.

The time-dependence and the elevated temperature necessary to attain rutile TiO₂ are distinct from critical factors reported in phase transitions for other TiO₂ synthetic methods. The significant differences in crystallisation temperatures for anatase and rutile phases offer the potential of controlling substrates for required phases type and ratio with post-deposition treatments.

Conclusions

This paper details the successful deposition of RF-sputtered TiO₂ films onto Si (110) substrates from TiO₂ targets with a range of deposition processes. The chemical composition, particularly with regards to the presence of contaminant species undesirable in surface-sensitive photoelectron studies was determined for a range of different deposition conditions. The results of these studies indicated that for minimal contaminant surface species the use of 30 minutes of pre-cleaning of the sputter target and the use of a new, 'fresh' TiO₂ target produced the surface with lowest concentrations of Si (<1%) and C (6%) achieved through heating to 500 °C under vacuum. These films are capable of modification through heating to induce Ti³⁺ defect sites. Morphological studies revealed that these films consisted of close-packed, films of circular nanoparticles 20–30 nm in size. Heating of the films increased the surface roughness and particle size of the TiO₂. The crystallinity of the substrate is capable of being tuned for anatase or an anatase/rutile mixture with heating between 500–1100 °C for extended periods (18–24 h).

Conflicts of interest

There are no conflicts to declare.

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