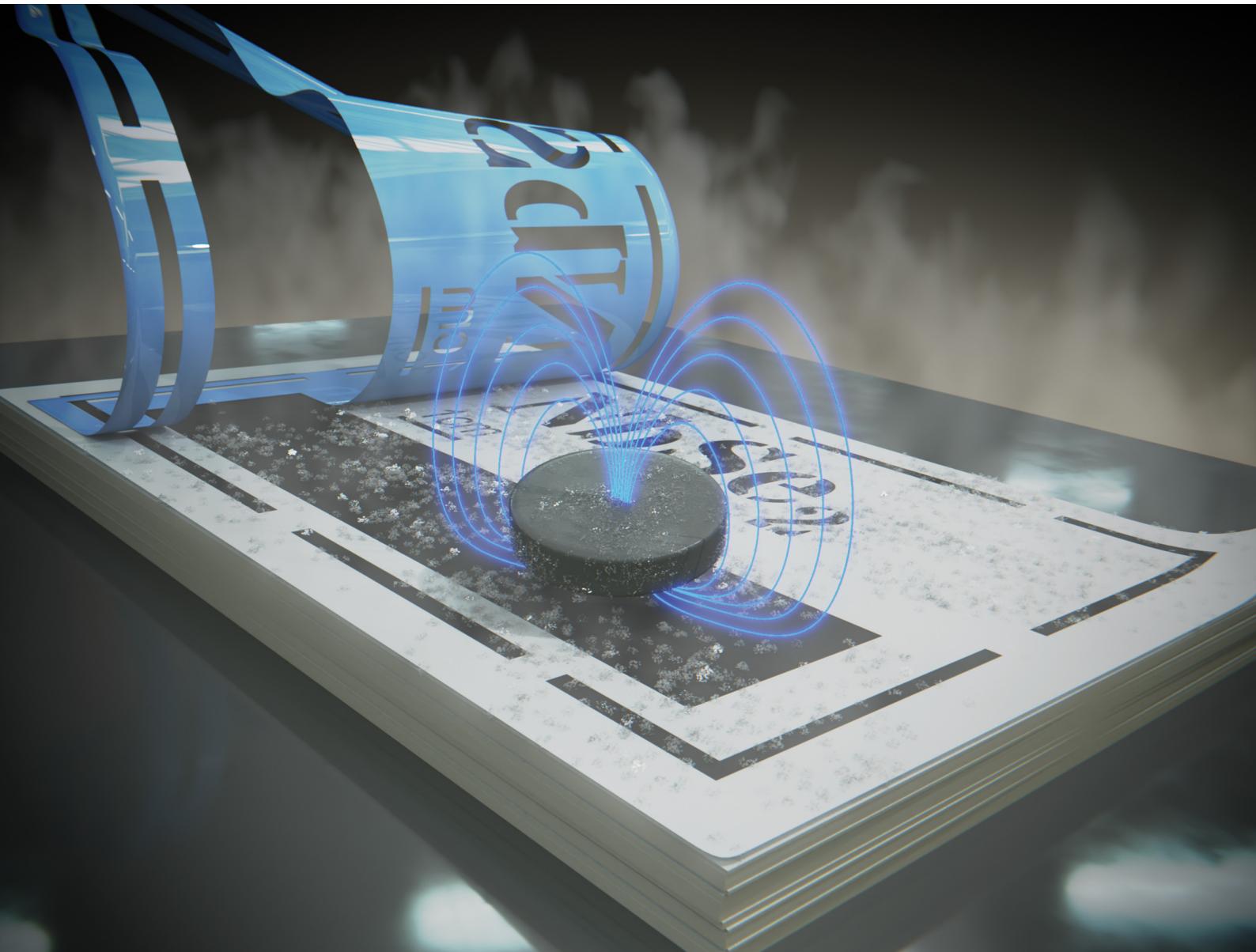


Materials Advances

rsc.li/materials-advances



ISSN 2633-5409

PAPER

Riccardo Frisenda, Mar García-Hernández,
Andrés Castellanos-Gómez *et al.*
Integrating superconducting van der Waals materials
on paper substrates



Cite this: *Mater. Adv.*, 2021, 2, 3274

Received 8th February 2021,
Accepted 5th April 2021

DOI: 10.1039/d1ma00118c

rsc.li/materials-advances

Introduction

The combination of low cost ($\sim 0.1 \text{ € m}^{-2}$), biodegradability and flexibility makes paper-based electronics very promising for applications like disposable wearable electronics and sensors.^{1–10} However, the integration of novel electronic materials, like the van der Waals (vdW) materials family, on paper substrates is hampered by their fibrous structure that introduces a large surface roughness and liquid-absorption. For these reasons, standard device fabrication approaches, developed and optimized for the fabrication of devices on silicon substrates, cannot be directly used on paper substrates. New fabrication techniques have been developed in the last years to overcome that issue, and now vdW materials can be deposited onto paper substrates using inkjet printing of inks prepared by liquid-phase exfoliation^{7,11–19} or by a recently reported all-dry abrasion-induced deposition method.^{20,21} Up to now these methods already demonstrated that one can fabricate heterostructures and pattern complex devices, with high spatial resolution, with conducting, semiconducting and insulating

vdW materials on paper.^{13–16} During the elaboration of this manuscript Novoselov, Lu and co-workers demonstrated the preparation of inks of monolayer NbSe₂ through a mild electrochemical exfoliation method and their use to print superconducting films on SiO₂/Si substrates by inkjet printing.²² The integration of vdW superconductors on paper substrates, however, is still lacking and its experimental realization is the goal of this manuscript.

Experimental

We deposited NbSe₂, as an illustrative example of vdW superconductor, on paper substrates by simply rubbing NbSe₂ fine powder ($\geq 99.8\%$, average particle size of 5 μm , Alfa Aesar PN: 13101.09) against the surface of a piece of standard copy paper with a cotton swab. During the rubbing process the flakes are subjected to friction forces that abrade the vdW platelet crystals exfoliating them and leading to the deposition of a dense network of interconnected flakes.^{20,21} The abrasion-induced deposition is carried out at atmospheric conditions until forming a homogeneous dark gray/black film. At this point we test the electrical continuity of the film with a handheld multimeter. A good electrically continuous film typically yields a resistance below 1 k Ω between two probes separated $\sim 1\text{--}2 \text{ mm}$. We have determined the sheet resistance of two films yielding values between $100 \Omega/\square^{-1}$ and $300 \Omega/\square^{-1}$ (See the ESI† Fig. S1). We address the reader to Ref. [23] for more detailed discussion about the resistivity measurements of abrasion-deposited van der Waals materials on paper.

^a Materials Science Factory, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid, E-28049, Spain. E-mail: riccardo.frisenda@csic.es, marmar@icmm.csic.es, andres.castellanos@csic.es

^b Kavli Institute of Nanoscience, Delft University of Technology, Lorentzweg 1, Delft, The Netherlands

† Electronic supplementary information (ESI) available: Supporting information includes further SEM/EDX characterization of the source NbSe₂ material and NbSe₂ film on paper, further details of the random resistor network model. See DOI: 10.1039/d1ma00118c



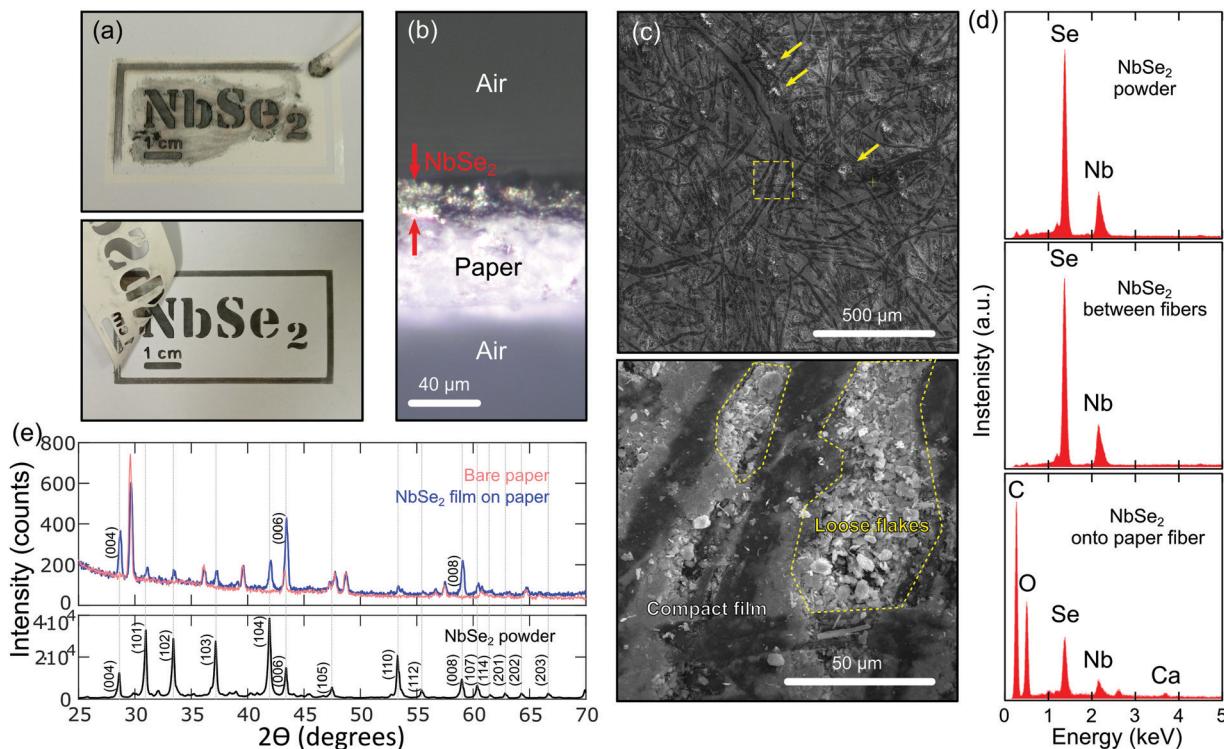


Fig. 1 Deposition of a NbSe₂ film on standard copy paper. (a) Pictures acquired during the deposition process of a NbSe₂ film on paper following a user-defined pattern thanks to a vinyl stencil mask. (b) Optical microscopy image of a cross-section of the paper coated with a NbSe₂ film where the thickness of the NbSe₂ can be estimated. (c) SEM images showing the morphology of the NbSe₂ film on paper. The bottom panel shows a zoomed-in image of the region highlighted with a dashed yellow square in the top panel, displaying the different morphologies between the material deposited onto the outermost paper fibers and that deposited filling in the gaps between fibers. (d) EDX analysis of the chemical composition of the source NbSe₂ powder material, the NbSe₂ deposited in the gap between the paper fibers and that deposited onto the outermost paper fibers. In the spectrum acquired on the NbSe₂ film deposited onto the paper fibers the signal of oxygen, carbon and calcium, originated by the paper substrate, arises pointing towards a lower thickness of the NbSe₂ film.

Fig. 1a shows pictures acquired during the deposition process of a NbSe₂ film on paper. A desktop vinyl cutter (Portrait Silhouette) has been used to fabricate a custom-made stencil mask that allows one to control the geometry of the deposited films with accuracy. By slicing the film with a sharp razor blade, we can even image the cross-section to estimate the average thickness of the film (see Fig. 1b) which is typically around 20 μm.

Results and discussion

We have characterized the morphology and the chemical composition of the deposited NbSe₂ film on paper by scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDX) using a FEI Helios G4 CX system. Fig. 1c show SEM images with low (top panel) and high (bottom panel) magnification of the NbSe₂ film on paper. The NbSe₂ is deposited over the whole surface of the paper except for few spots of uncoated or barely coated paper that show up in the low magnification SEM image as brighter spots (highlighted with yellow arrows in Fig. 1c top panel). From the SEM images we infer that abrasion during the deposition process crushes the NbSe₂ flakes forming

a rather compact film. Inside the gaps between the paper fibers, however, it is sometimes possible to resolve NbSe₂ flakes with lateral sizes in the 1–5 μm range (slightly smaller than lateral dimensions of the flakes in the source NbSe₂ powder, see Fig. S2, ESI†). EDX spectroscopy analysis provides an insight about the chemical composition of the as-drawn NbSe₂ film. Fig. 1d compares EDX spectra acquired on the source NbSe₂ powder, on the NbSe₂ deposited filling in the gaps between paper fibers and on the NbSe₂ deposited onto the outermost paper fibers. While the EDX spectra of NbSe₂ powder and NbSe₂ deposited between fibers are comparable, the spectra acquired on top of the outermost paper fibers show signals of NbSe₂ together with signals of oxygen, carbon and calcium originating from the paper underneath. In fact, a calcium carbonate (a bright white mineral with chemical formula CaCO₃) is quite often added to paper pulp as a filler. This result indicates that the NbSe₂ film on top of the outermost paper fibers is rather thin. See Fig. S2 to S4 in the ESI† for further SEM and EDX analysis.

We further characterized the morphology of the samples by X-ray diffraction (XRD) to clarify if the NbSe₂ flakes have a preferential orientation induced by the deposition method. Fig. 1e compares the X-ray diffraction (XRD) patterns, acquired

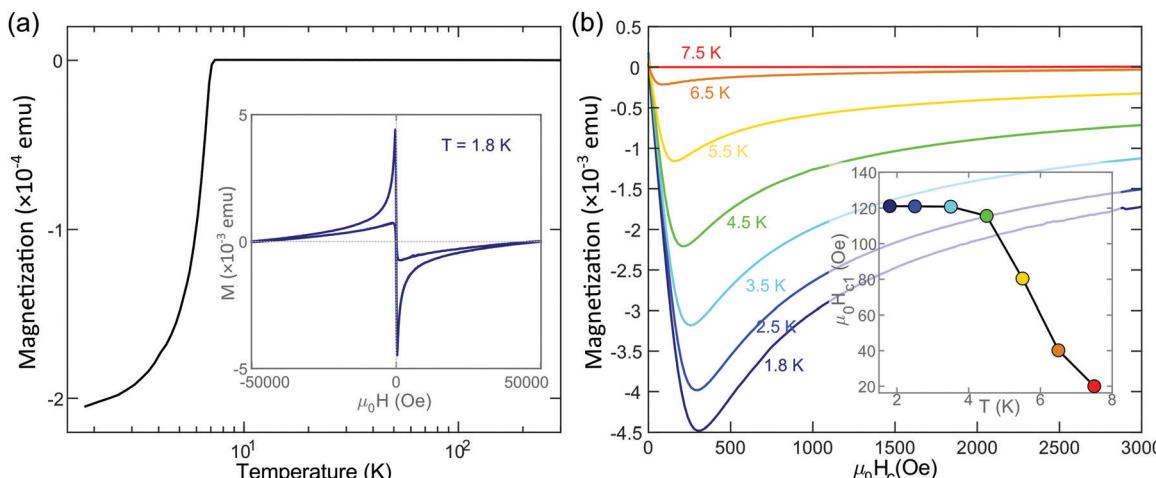


Fig. 2 Magnetization characterization of a NbSe_2 film on standard copy paper. (a) Magnetization vs. temperature, measured after a zero-field cooling using 10 Oe of measuring field (sample 1). (Inset in a) Magnetization vs. magnetic field cycle, starting at zero-field, measured on a NbSe_2 film on paper after subtracting the paper contribution (b) Close-up of the first-magnetization curve at several temperatures. (Inset in b) H_{c1} derived from the linear fits to the first-magnetization curve at low fields.

with Cu K α 1 radiation, on the source NbSe_2 powder, on the NbSe_2 film on paper and on the bare (uncoated) paper. The XRD pattern of NbSe_2 powder agrees well with the expected for bulk NbSe_2 . The XRD pattern of bare paper allows us to identify the features originated from the substrate in the XRD pattern measured on the NbSe_2 -on-paper film. From a direct comparison between the patterns acquired on the powder and on the film on paper one can see how on the film the (00n) diffraction peaks are systematically more intense than the other peaks, indicating a preferential alignment of the NbSe_2 flakes with their basal plane parallel to the paper surface.

In order to characterize the superconducting properties of the as-deposited NbSe_2 films we have first used a superconducting quantum interference device (SQUID) magnetometer (Quantum Design) equipped with a 5 Tesla coil. Fig. 2a shows the measured magnetization as a function of the temperature after a zero-field cooling, using a measuring field of 10 Oe in a direction parallel to the substrate and film. A clear transition is observed around $T \sim 7$ K that can be interpreted as the onset of the superconducting state exhibiting a strong and sudden decrease of the magnetization due to the Meissner effect. Another evidence of the superconducting transition comes from the isotherm curve measured at 1.8 K in the inset: the NbSe_2 -on-paper sample was cooled in zero-field from 300 K to 1.8 K and a magnetization *versus* magnetic field curve, starting from zero field, was measured. The inset in Fig. 2a shows the diamagnetic response of the NbSe_2 sample on paper after subtraction of the paper contribution. Note the diamond-like shape of the cycle pointing out to the typical magnetic response of a superconductor below its critical temperature T_c . Fig. 2b shows a close-up of the isothermal first-magnetization curves for various temperatures. Linear fits to the curves at low field render the first critical field values H_{c1} of a typical type II superconductor (shown in the inset of Fig. 2b). H_{c2} is larger than the 5 Tesla accessible with our SQUID coil.

We have also studied the temperature dependence of the resistance of a $\sim 5 \times 5 \text{ mm}^2$ NbSe_2 film on paper with the van der Pauw configuration in a physical property measurement system (PPMS, by Quantum Design) cryostat equipped with a 9 Tesla coil (with the magnetic field perpendicular to the film and paper substrate). Fig. 3 shows the resulting resistance *vs.* temperature (R *vs.* T hereafter) where there is an overall increase of the resistance upon sample cooldown. This comes to a surprise since metallic character has been reported for NbSe_2 and thus one would expect a continuous decrease of the resistance when decreasing the temperature. We attribute this

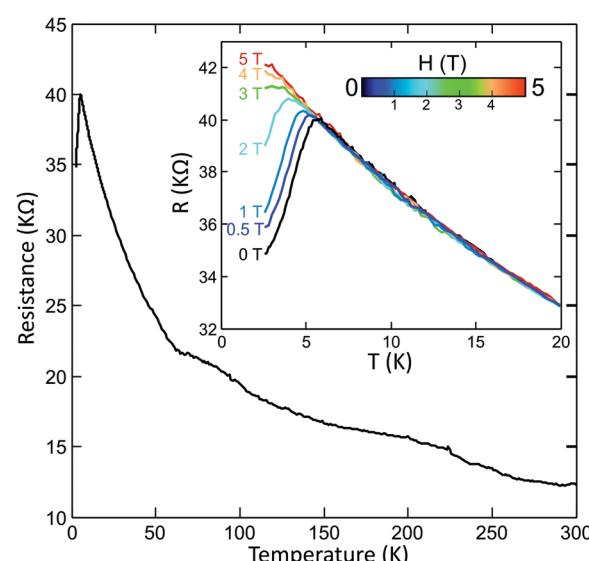


Fig. 3 Temperature dependent resistance of a NbSe_2 film on standard copy paper. (a) Resistance *vs.* temperature measured on a NbSe_2 -on-paper film at zero-field (sample 3, van der Pauw geometry). (Inset) Resistance *vs.* temperature curves measured at different applied magnetic fields.



behavior to localization due to the disordered nature of the film which can be described in terms of a thermally activated hopping mechanism between the interconnected flakes of the film as detailed later. Several thermally activated transport mechanisms, such as Arrhenius-like or variable range hopping, have yielded similar temperature dependences in other percolative systems composed of highly-conductive particles with a highly-resistive particle-to-particle interface.^{24,25}

No specific feature linked to the onset of a charge density wave transition, with a reported transition temperature around $T = 145$ K and 35 K for single-layer and bulk samples respectively, has been observed in Fig. 3.^{26,27} However, several jumps of the resistivity while decreasing the temperature are apparent. Interestingly, at $T \sim 6$ K the R vs. T changes the overall trend and the resistance drops rapidly upon cooldown. This turning point matches well with the critical temperature observed in the SQUID measurement and points to the transition of the NbSe_2 flakes into the superconducting state. The high degree of disorder in the film and the interflake hopping resistance, however, prevents from observing a sharp electrical transition to a zero-resistance state. In fact, here we recall the polycrystalline nature of our superconducting film, with a broad distribution of flakes with various thickness, which could exhibit different critical temperatures,^{28–33} and a variety of lateral sizes. Note that the large residual resistance below the superconducting transition is expected due to the large film channel length (~ 5 mm), orders of magnitude larger than the typical NbSe_2 platelets size, as the current has to pass through several interflake junctions that behave as a normal-state resistor. We anticipate that shorter channel devices will present a sharper transition with a reduced residual resistance. In fact recent results, reported during the elaboration of this work, on inkjet printed films of interconnected NbSe_2 layers on SiO_2/Si

substrates show a very abrupt superconducting transition with negligible residual resistance with electrode separation of $<20 \mu\text{m}$.²² Inkjet printing, however, cannot be used for printing devices on standard copy paper, since the low-viscosity nanoinks tend to leak through the cellulose fibers.

The observed behaviour in the film of NbSe_2 flakes on paper can be qualitatively explained by a model of a random resistor network of interconnected NbSe_2 flakes. A small network of 4×4 resistors is shown, as a simplified example, in Fig. 4a. The network is composed of two different kind of resistors called SC and I, which respectively represent the resistance of the superconducting NbSe_2 flakes (SC) and the flake-to-flake hopping (I) as depicted in Fig. 4a. To populate the random resistor network, we define a probability parameter p ($0 < p < 1$), which controls the percentage of I and SC resistors in the network. For each resistor we generate a random number x uniformly distributed between 0 and 1. If $x < p$ we assign the resistor to the state I and if $x \geq p$ to SC. The simulations presented in the main text are performed using $p = 0.65$, a value that ensures a configuration with small clusters of SC resistors surrounded by I resistors. This configuration emulates the NbSe_2 platelets (SC islands) connected through flake-to-flake junctions (I resistors), see Fig. 4a. The network can be biased by applying a voltage to the nodes in the network, in the 4×4 example the left side is kept at 1 V (red circles) and the right side at 0 V (black circles). After solving for the unknown voltages (white circles) one can calculate the current flowing through each resistor and the total current in the network using Ohm's law. See ESI† Fig. S5 to S7 for more details about the random resistor network model and for simulations with different critical temperature and probability p parameters.

Fig. 4b shows the temperature dependence of the SC and I resistances used for the simulation. We have considered a

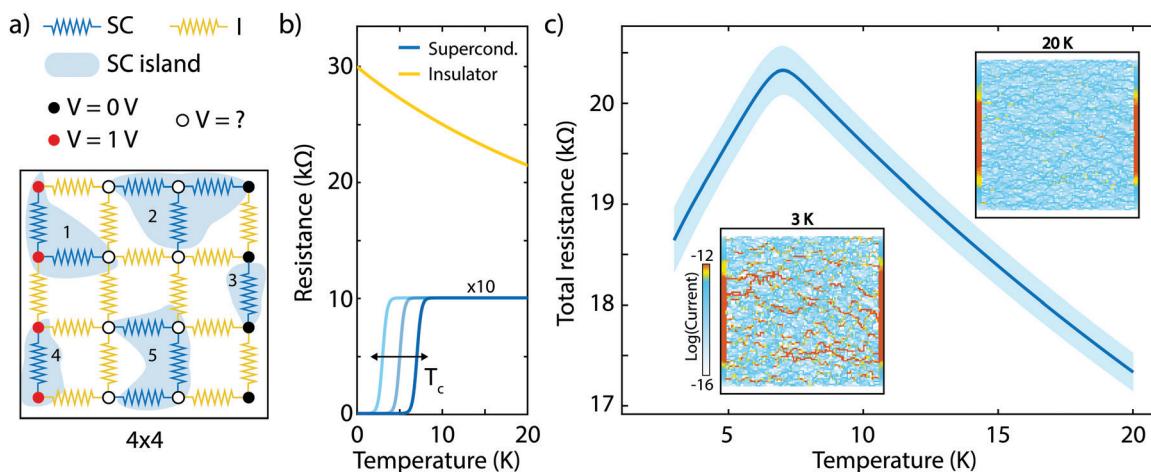


Fig. 4 Modelling the superconducting transition in a percolative film. (a) Schematic of the random resistor network used to simulate the behavior of the interconnected network of NbSe_2 flakes. A small network of 4×4 resistors is shown as a simplified example. The network is composed by different resistors SC and I that represent the superconducting NbSe_2 flakes and the flake-to-flake hopping resistance respectively. (b) Temperature dependence of the T_c of the SC resistors from 1 K to 7 K to account for the thickness dependent T_c of NbSe_2 . The I resistors are modelled as a thermally activated hopping mechanism. (c) Mean total resistance calculated from 10 different 120×120 random resistors networks as a function of the temperature. The light blue bands correspond to a variation of 1 standard deviation from the mean curve. The insets show the current magnitude maps calculated in the normal (20 K) and the superconducting (3 K) state.



variation of the critical temperature T_c of the SC resistors from 1 K to 7 K to account for the thickness dependent T_c of NbSe_2 .^{28–33} The I resistors are modelled as a thermally activated resistors. Fig. 4c shows the mean resistance calculated from 10 different 120×120 random resistor networks as a function of the temperature, which shows a decrease of resistance at ~ 6 K without dropping all the way to zero Ohms, a behaviour similar to that observed experimentally. The observation of a residual resistance below T_c can be explained by the presence of the inter-platelet normal-state resistances, which are connected in series to the superconducting elements and thus limit the supercurrent flow. The model also reproduces the overall resistance increase upon cooling down for temperatures above T_c . The insets show the current magnitude maps calculated in the normal (20 K) and the superconducting (3 K) state. At 3 K highly conductive filaments (red filaments in the inset in Fig. 4c) arise because of the transition of the SC resistors to the low resistance state, reducing the mean resistance of the network. Note that no filaments fully composed of SC resistors bridge the two electrodes. Our model also predicts that shorter channel devices will present more abrupt superconducting transitions and with a lower residual resistance below the superconducting transition, as discussed above. See the ESI† Fig. S8 for the results of simulations of resistor networks with shorter and shorter channel lengths where this trend is shown.

We have further studied the magnetic field dependence of the NbSe_2 film on paper resistance (e.g., the inset in Fig. 3 shows the magnetic field dependence of the R vs. T curve around T_c). As expected for a superconducting transition, as the magnetic field increases the critical temperature is depleted to a point such that above 3 Tesla the transition is basically suppressed in the explored temperature range. Fig. 5a shows the

magnetic field dependence of the NbSe_2 film on paper resistance (R vs. H) at different temperatures. Above T_c the system exhibits a negative magnetoresistance and the resistance decreases as an external magnetic field is applied. Below T_c , the magnetoresistance changes sign and the resistance increases dramatically when the magnetic field is applied. This behavior can be rationalized in terms of the model proposed by Porat *et al.* for highly disordered superconductors.³⁴ Again, the rationale behind is that a highly disordered superconductor can be viewed as a set of superconducting percolating paths. As a magnetic field is applied the smaller or less robust superconducting flakes enter into the normal state and the number of superconducting paths decreases and thus the resistance increases. As the field is further increased, the conduction evolves to a point where the superconductor paths are not connected and thus the normal current paths become dominant. The inset in Fig. 5a shows the applied magnetic field when the normal state sets in for the measured temperatures. This could be interpreted as the temperature dependence of an effective upper critical field H_{c2} of the film.

We have also measured the angle dependence of the magnetoresistance of another NbSe_2 film on paper, with a bar-shape ($5 \times 2 \text{ mm}^2$) and probed with four-terminal sensing geometry. Comparing resistance measurements at two-terminal and four-terminal sensing we can infer that the contact resistance ranges from $\sim 10 \text{ k}\Omega$ at room temperature to $\sim 500 \text{ k}\Omega$ at ~ 4 K. We used another PPMS (Quantum Design) cryostat equipped with a 14 Tesla coil (with the magnetic field perpendicular to the film and paper substrate) and a rotator mount for the sample. Fig. 5b shows the magnetic field dependence of the NbSe_2 film resistance, at 5 K, for several angles ranging from -90° and 90° . The angle is labelled as -90° and 90° for

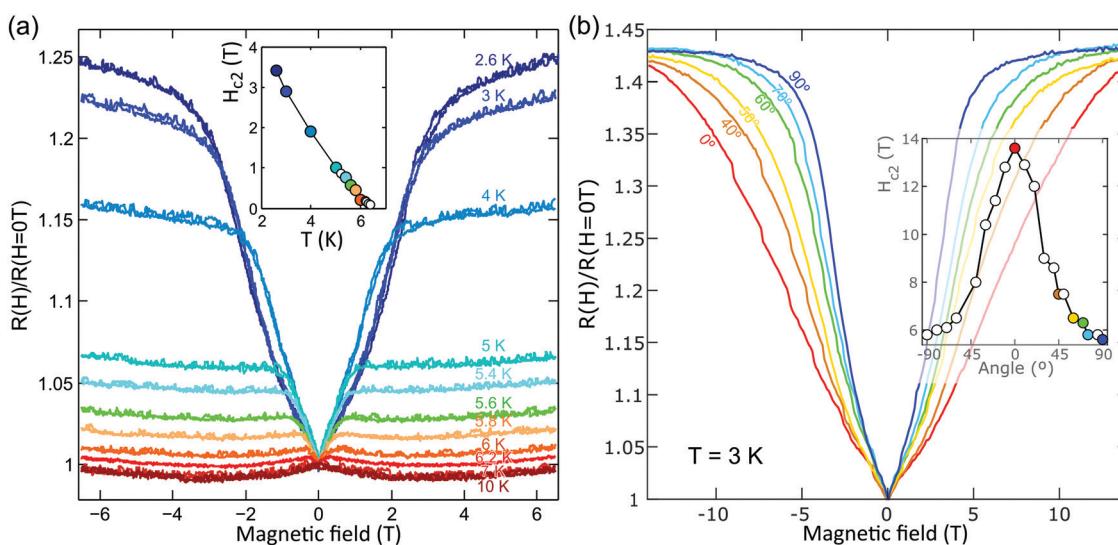


Fig. 5 Magneto-transport in a NbSe_2 film on paper at different temperatures and magnetic field rotation. (a) Resistance vs. magnetic field, measured at different temperatures around the superconducting transition temperature (sample 4, four-terminal geometry). (Inset in a) Upper critical field, H_{c2} , values extracted for different temperatures from the magneto-transport data in (a). (b) Resistance of the NbSe_2 film vs. Magnetic field, measured for different magnetic field tilting angles ranging from: perpendicular (-90° and 90°) to parallel (0°) to the paper surface. (Inset in b) Ratio between the upper critical field H_{c2} measured at a given magnetic field tilting angle (θ) and that measured with the field perpendicular to the paper surface.



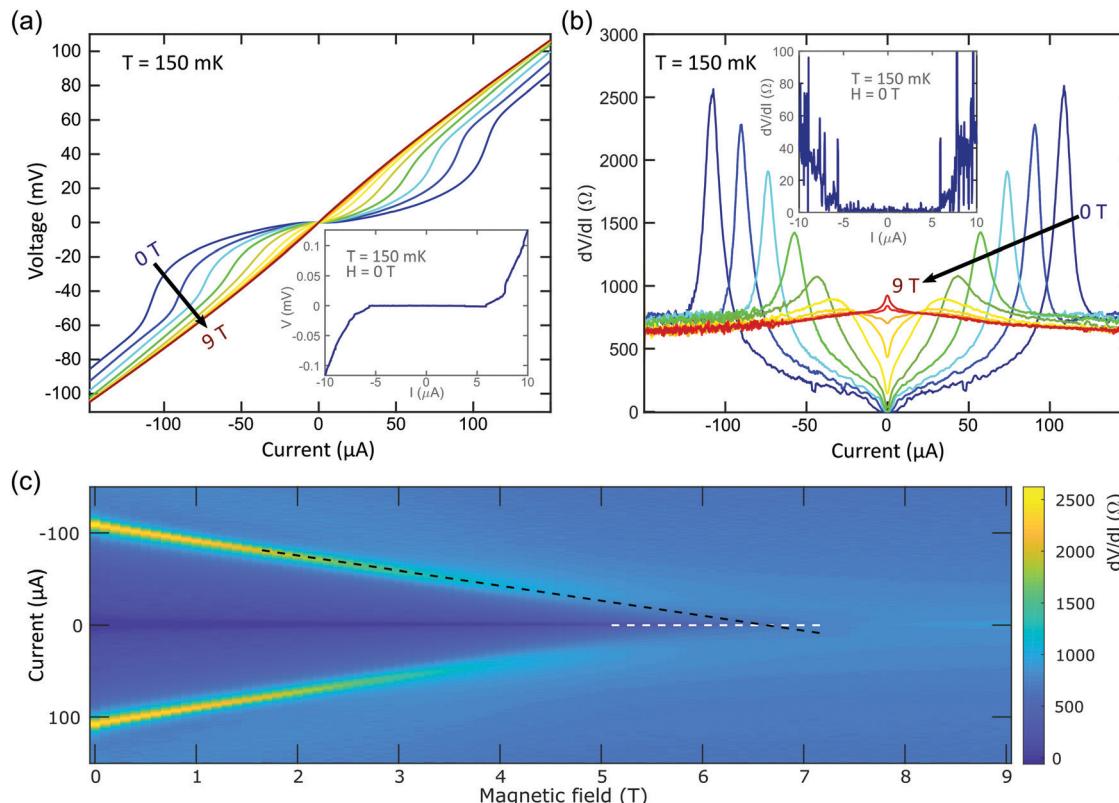


Fig. 6 Magneto-transport in a NbSe₂ film on paper at 150 mK. (a) Voltage vs. current curves, measured at different magnetic fields (sample 5, van der Pauw geometry). (Inset in a) Detail of the voltage vs. current curve at $H = 0$ T which displays a zero-resistance state between $\sim \pm 5$ μ A current bias. (b) Derivative of the voltage vs. current curves (dV/dI) at different fields (from 0 T to 9 T, in 1 T steps) where the destruction of the zero-resistance state at high magnetic field is evident. (c) False color map of the dV/dI (in the color axis) as a function of the bias current (vertical axis) and magnetic field (horizontal axis). The superconductivity is destroyed at $H \sim 6.5$ T.

magnetic fields perpendicular to the film, H_{\perp} , and as 0° for magnetic field parallel to the film, H_{\parallel} . The inset in Fig. 5b plots the upper critical field H_{c2} values extracted for different magnetic field tilting angles, displaying a huge anisotropy. Similar anisotropy has been also observed in single crystal NbSe₂, in agreement with the preferential alignment of the NbSe₂ platelets with their basal plane parallel to the paper surface (see the XRD data in Fig. 1c).³³

To gain a deeper insight into the electrical transport characteristics of the NbSe₂ films on paper below the superconducting critical temperature we have performed further magnetotransport measurements with a dilution fridge in the 100 mK to 1000 mK temperature range. Voltage vs. current curves have been measured at different magnetic fields (perpendicular to the film) on another NbSe₂ film on paper device ($\sim 5 \times 5$ mm² with van der Pauw geometry, see Fig. 6a). For fields below 7 T, the curves display a zero-resistance state indicating fully developed superconducting transport across the sample. We have found that the electrical characteristics of the film do not change substantially in the studied temperature range. Fig. 6b shows the derivative of the voltage vs. current traces (dV/dI) which facilitate the observation of the zero-resistance states at low current biasing. Fig. 6c summarizes the dV/dI as a function of the magnetic field and the current

bias conditions through a false color map where it is shown how the superconductivity is completely suppressed by magnetic fields higher than ~ 6.5 T.

Conclusions

In summary, we show a facile route to integrate van der Waals superconductors on standard copy paper substrates by simply rubbing powder of the selected superconductor against paper. We illustrate it with NbSe₂, a prototypical vdW superconductor, finding that the as-deposited NbSe₂ films expel magnetic fields at temperatures lower than 7 K, a clear proof that they display Meissner effect. The resistance of NbSe₂ on paper also shows an abrupt drop at temperature lower than 6 K, consistent with a superconducting transition of the NbSe₂ flakes composing the interconnected network of platelets that is the film. This behavior can be accurately reproduced with a random resistor network model. Furthermore, magnetotransport measurements carried out at 150 mK show a complete superconducting transition, displaying a zero-resistance state across the sample (at millimeters scale), with a critical field of ~ 6.5 T. The results shown here are robust (all the 5 studied NbSe₂-on-paper samples present a superconducting transition) and general

and thus they open the door to integrate other vdW superconductors on standard paper substrates. We believe that, given the low cost and low weight of paper substrates, this technique can become a new route towards mass-scalable production of simple superconducting devices (like superconducting high frequency filters) or as coatings for magnetic field shielding in cryogenic applications.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This project has received funding from the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation program (grant agreement n° 755655, ERC-StG 2017 project 2D-TOPSENSE) and the European Union's Horizon 2020 research and innovation program under the Graphene Flagship (grant agreement number 785219, GrapheneCore2 project and grant agreement number 881603, GrapheneCore3 project). R. F. acknowledges the support from the Spanish Ministry of Economy, Industry and Competitiveness (MINECO) through a Juan de la Cierva-formación fellowship 2017 FJCI-2017-32919 and the grant MAT2017-87134-C2-2-R. This work was supported by the Netherlands Organisation for Scientific Research (NWO/OCW), as part of the Frontiers of Nanoscience program. W. Zhang acknowledges the grant from China Scholarship Council (CSC) under No. 201908610178.

Notes and references

- 1 D. Tobjörk and R. Österbacka, *Adv. Mater.*, 2011, **23**, 1935–1961.
- 2 A. Russo, B. Y. Ahn, J. J. Adams, E. B. Duoss, J. T. Bernhard and J. A. Lewis, *Adv. Mater.*, 2011, **23**, 3426–3430.
- 3 N. Kurra and G. U. Kulkarni, *Lab Chip*, 2013, **13**, 2866–2873.
- 4 G. Zheng, Y. Cui, E. Karabulut, L. Wågberg, H. Zhu and L. Hu, *MRS Bull.*, 2013, **38**, 320–325.
- 5 C. Casiraghi, M. Macucci, K. Parvez, R. Worsley, Y. Shin, F. Bronte, C. Borri, M. Paggi and G. Fiori, *Carbon*, 2018, **129**, 462–467.
- 6 J. Liu, C. Yang, H. Wu, Z. Lin, Z. Zhang, R. Wang, B. Li, F. Kang, L. Shi and C. P. Wong, *Energy Environ. Sci.*, 2014, **7**, 3674–3682.
- 7 S. Conti, L. Pimpolari, G. Calabrese, R. Worsley, S. Majee, D. K. Polyushkin, M. Paur, S. Pace, D. H. Keum, F. Fabbri, G. Iannaccone, M. Macucci, C. Coletti, T. Mueller, C. Casiraghi and G. Fiori, *Nat. Commun.*, 2020, **11**, 3566.
- 8 P. T. Gomathi, P. Sahatiya and S. Badhulika, *Adv. Funct. Mater.*, 2017, **27**, 1701611.
- 9 L. Gao, C. Zhu, L. Li, C. Zhang, J. Liu, H.-D. Yu and W. Huang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 25034–25042.
- 10 L. Gao, L. Chao, M. Hou, J. Liang, Y. Chen, H.-D. Yu and W. Huang, *npj Flexible Electron.*, 2019, **3**, 4.
- 11 D. McManus, S. Vranic, F. Withers, V. Sanchez-Romaguera, M. Macucci, H. Yang, R. Sorrentino, K. Parvez, S.-K. Son, G. Iannaccone, K. Kostarelos, G. Fiori and C. Casiraghi, *Nat. Nanotechnol.*, 2017, **12**, 343.
- 12 C. Zhang, L. McKeon, M. P. Kremer, S.-H. Park, O. Ronan, A. Seral-Ascaso, S. Barwick, C. Ó. Coileáin, N. McEvoy, H. C. Nerl, B. Anasori, J. N. Coleman, Y. Gogotsi and V. Nicolosi, *Nat. Commun.*, 2019, **10**, 1795.
- 13 J. Li, M. M. Naiini, S. Vaziri, M. C. Lemme and M. Östling, *Adv. Funct. Mater.*, 2014, **24**, 6524–6531.
- 14 D. J. Finn, M. Lotya, G. Cunningham, R. J. Smith, D. McCloskey, J. F. Donegan and J. N. Coleman, *J. Mater. Chem. C*, 2014, **2**, 925–932.
- 15 T. Carey, S. Cacovich, G. Divitini, J. Ren, A. Mansouri, J. M. Kim, C. Wang, C. Ducati, R. Sordan and F. Torrisi, *Nat. Commun.*, 2017, **8**, 1202.
- 16 A. G. Kelly, T. Hallam, C. Backes, A. Harvey, A. S. Esmaeily, I. Godwin, J. Coelho, V. Nicolosi, J. Lauth, A. Kulkarni, S. Kinge, L. D. A. Siebbeles, G. S. Duesberg and J. N. Coleman, *Science*, 2017, **356**, 69–73.
- 17 G. Cunningham, M. Lotya, N. McEvoy, G. S. Duesberg, P. van der Schoot and J. N. Coleman, *Nanoscale*, 2012, **4**, 6260–6264.
- 18 A. G. Kelly, C. Murphy, V. Vega-Mayoral, A. Harvey, A. S. Esmaeily, T. Hallam, D. McCloskey and J. N. Coleman, *2D Mater.*, 2017, **4**, 41006.
- 19 J. Li, M. M. Naiini, S. Vaziri, M. C. Lemme and M. Östling, *Adv. Funct. Mater.*, 2014, **24**, 6524–6531.
- 20 D. Nutting, J. F. Felix, E. Tillotson, D.-W. Shin, A. De Sanctis, H. Chang, N. Cole, S. Russo, A. Woodgate, I. Leontis, H. A. Fernández, M. F. Craciun, S. J. Haigh and F. Withers, *Nat. Commun.*, 2020, **11**, 3047.
- 21 A. Mazaheri, M. Lee, H. S. J. van der Zant, R. Frisenda, A. Castellanos-Gomez, M. Lee, H. S. J. van der Zant, R. Frisenda and A. Castellanos-Gomez, *Nanoscale*, 2020, **12**, 19068–19074.
- 22 J. Li, P. Song, J. Zhao, K. Vaklinova, X. Zhao, Z. Li, Z. Qiu, Z. Wang, L. Lin, M. Zhao, T. S. Herring, Y. Zuo, W. Jonhson, W. Yu, X. Hai, P. Lyu, H. Xu, H. Yang, C. Chen, S. J. Pennycook, J. Ding, J. Teng, A. H. Castro Neto, K. S. Novoselov and J. Lu, *Nat. Mater.*, 2021, **20**, 181–187.
- 23 W. Zhang, Q. Zhao, C. Munuera, M. Lee, E. Flores, J. E. F. Rodrigues, J. R. Ares, C. Sanchez, J. Gainza, H. S. J. van der Zant, J. A. Alonso, I. J. Ferrer, T. Wang, R. Frisenda and A. Castellanos-Gomez, *Appl. Mater. Today*, 2021, **23**, 101012.
- 24 J. Halim, E. J. Moon, P. Eklund, J. Rosen, M. W. Barsoum and T. Ouisse, *Phys. Rev. B*, 2018, **98**, 104202.
- 25 K.-H. Müller, J. Herrmann, B. Raguse, G. Baxter and T. Reda, *Phys. Rev. B*, 2002, **66**, 075417.
- 26 X. Xi, L. Zhao, Z. Wang, H. Berger, L. Forró, J. Shan and K. F. Mak, *Nat. Nanotechnol.*, 2015, **10**, 765–769.
- 27 M. M. Ugeda, A. J. Bradley, Y. Zhang, S. Onishi, Y. Chen, W. Ruan, C. Ojeda-Aristizabal, H. Ryu, M. T. Edmonds, H.-Z. Tsai, A. Riss, S.-K. Mo, D. Lee, A. Zettl, Z. Hussain, Z.-X. Shen and M. F. Crommie, *Nat. Phys.*, 2016, **12**, 92–97.



28 R. F. Frindt, *Phys. Rev. Lett.*, 1972, **28**, 299–301.

29 N. E. Staley, J. Wu, P. Eklund, Y. Liu, L. Li and Z. Xu, *Phys. Rev. B*, 2009, **80**, 184505.

30 M. S. El-Bana, D. Wolverson, S. Russo, G. Balakrishnan, D. M. Paul and S. J. Bending, *Supercond. Sci. Technol.*, 2013, **26**, 125020.

31 T. Dvir, F. Masse, L. Attias, M. Khodas, M. Aprili, C. H. L. Quay and H. Steinberg, *Nat. Commun.*, 2018, **9**, 598.

32 E. Khestanova, J. Birkbeck, M. Zhu, Y. Cao, G. L. Yu, D. Ghazaryan, J. Yin, H. Berger, L. Forró, T. Taniguchi, K. Watanabe, R. V. Gorbachev, A. Mishchenko, A. K. Geim and I. V. Grigorieva, *Nano Lett.*, 2018, **18**, 2623–2629.

33 N. Toyota, H. Nakatsuji, K. Noto, A. Hoshi, N. Kobayashi, Y. Muto and Y. Onodera, *J. Low Temp. Phys.*, 1976, **25**, 485–499.

34 E. Porat and Y. Meir, *Phys. Rev. B*, 2015, **92**, 024509.

