



First synthesis of air-stable NiZn homogeneous alloy nanoparticles through chemical reduction†

Cite this: *Mater. Adv.*, 2021, 2, 684

Received 13th November 2020,
Accepted 19th December 2020

DOI: 10.1039/d0ma00883d

rsc.li/materials-advances

Shinya Okamoto,^{ib}*^a Kohei Kusada,^{ib}^b Hiroshi Hashiba,^a Satoshi Yotsuhashi^a and Hiroshi Kitagawa^{ib}^b

We successfully synthesized NiZn homogeneous alloy nanoparticles, which were mixed at the atomic level, for the first time through a chemical reduction method with sodium–naphthalenide as a reducing agent. The NiZn alloy nanoparticles were stable even in air for at least half a year, although Zn is typically easily oxidized.

Alloy nanoparticles have received considerable attention owing to the tunability of their chemical and physical properties based on control of their composition,^{1–5} particle size, shape,^{6–10} and mixing pattern.^{11,12} In addition, it has been reported that new types of alloy nanoparticles, such as AgRh¹³ and PdRu¹⁴ solid-solution alloy nanoparticles, have been synthesized by a chemical reduction process despite immiscibility of the constituent elements.^{15–19} Syntheses of new alloy nanoparticles give insights into materials science. For example, AgRh nanoparticles have hydrogen-storage ability, although neither Ag nor Rh show this property. PdRu nanoparticles also have a highly efficient CO-oxidizing catalytic activity. However, there remain challenges to establishing robust approaches to synthesizing alloy nanoparticles from arbitrary elements. Most reported alloy nanoparticles contain noble metals because noble metal nanoparticles are easier to be synthesized than non-noble metals through chemical methods owing to their relatively high electronegativities. Therefore, there have been few reports on alloy nanoparticles consisting of only non-noble metals synthesized by chemical methods.

In this report, we focus on a combination of the non-noble metals, Ni and Zn. There have been several reports on catalysts containing Ni and Zn. For example, Ni and Zn dispersed on an oxide support by an incipient wetness technique were capable

of selective hydrogenation of acetylene.²⁰ Nanoparticles synthesized from Ni/ZnO-coated carbon nanotubes gave selective conversion of cellulose into vicinal diols.²¹ Another study found that nanoparticles synthesized from Ni(NO₃)₂·6H₂O and Zn(NO₃)₂·6H₂O by a surfactant-aided polyol reduction method were an efficient catalyst for phenol gasification.²² These reports highlight the great potential of NiZn alloy nanoparticles as efficient catalysts. However, to the best of our knowledge, there have been no reports of uniformly mixed alloy nanoparticles of NiZn. The most successful result of this challenge was the synthesis of nanoparticles having a core–shell structure with Zn atoms in an Ni core by Jana *et al.*²³ The difficulty in the synthesis of homogeneous NiZn alloy nanoparticles is attributed to the large differences in the chemical properties between Ni and Zn, such as their melting points (Ni: 1455 and Zn: 420 °C),²⁴ electron negativities, and ease of oxidation in air.

Here, we demonstrate a chemical synthesis of homogeneous NiZn alloy nanoparticles with the use of sodium–naphthalenide as a reducing agent. Sodium–naphthalenide has a large reduction capability, and it has recently been used in the synthesis of Zn nanoparticles.^{25–27} The crystal structure was analyzed by synchrotron powder X-ray diffraction (XRD), and the detailed structure analysis was implemented with transmission electron microscopy (TEM) and scanning TEM (STEM) coupled with energy-dispersive X-ray spectroscopy (EDX).

NiZn nanoparticles were synthesized under an argon atmosphere using Schlenk techniques at room temperature. First, sodium–naphthalenide solution was prepared. A flask containing naphthalene was filled with argon at room temperature. Then, tetrahydrofuran (THF) was injected by syringe, and cut sodium was dropped into the solvent. The solution was stirred for 12 h. Second, a metal precursor solution was prepared by dissolving NiCl₂·CH₃O(CH₂)₂OCH₃ (0.07 mmol) and anhydrous ZnCl₂ (0.14 mmol) in THF (17 mL). The color of the precursor solution quickly turned black on injection of the sodium–naphthalenide solution (5 mL), which indicated the formation of metal nanoparticles. After stirring for 5 min, the as-synthesized nanoparticles were removed to air and separated with a centrifuge at 210 000 *g*-force (50 000 rpm) for 2 h. Finally, the nanoparticles

^a Technology Division, Panasonic Corporation, 1006 Kadoma, Kadoma-City, Osaka 571-8508, Japan

^b Division of Chemistry, Graduate School of Science, Kyoto University, Kitashirakawa-Oiwakecho, Sakyo-ku, Kyoto 606-8502, Japan

† Electronic supplementary information (ESI) available: TEM image, Rietveld refinement of XRD pattern, STEM images of the surface, EDX data of synthesized NiZn nanoparticles, STEM-EDX maps and XRD pattern of phase-separated nanoparticles. See DOI: 10.1039/d0ma00883d



were filtered with methanol and dried under vacuum. The dried powder was sintered at 300 °C for 1 h in hydrogen. The nanoparticles were obtained with yields of 90–95%.

We note the importance of drying the starting material for the Ni precursor. When H₂O is included in the Ni salt, it appears to lower the reduction ability of sodium-naphthalenide. NiCl₂·CH₃O(CH₂)₂OCH₃ provided the best results among several kinds of Ni precursors in this study. The solubility of the precursors in the solvent should also be carefully considered to obtain homogeneous alloy nanoparticles. Most commercial Ni compounds, such as Ni(NO₃)₂, Ni(OCOCH₃)₂, NiSO₄, and NiCO₃, were practically insoluble in THF. Moreover, the density of the sodium-naphthalenide should be adjusted to an appropriate level.

The synthesized nanoparticles were characterized by TEM imaging. Fig. S1 (ESI[†]) shows that the nanoparticles were synthesized by a sodium-naphthalenide reduction without any stabilizing agents.

The crystal structure of the synthesized nanoparticles was investigated with the use of synchrotron powder XRD at 303 K, at the beamline BL02B2, SPring-8.²⁸ Fig. 1 shows the XRD patterns of the synthesized NiZn nanoparticles and Ni and Zn powder. The diffraction patterns of Ni and Zn had face-centered cubic (fcc) and hexagonal close-packed (hcp) structures, respectively. The XRD pattern of the synthesized nanoparticles corresponds to β₁-phase of a NiZn intermetallic alloy with a tetragonal structure (Fig. S2, ESI[†]).²⁹ We note that peaks of the Ni, Zn mono-metals, or other by-products, such as ZnO were not observed in the patterns of the nanoparticles. This result confirms that the NiZn alloy nanoparticles were uniformly synthesized.

To investigate the detailed structure of the synthesized nanoparticles, we performed STEM observations. Fig. 2a shows a high-angle annular dark-field STEM (HAADF-STEM) image of the synthesized nanoparticles. Fig. 2b and c show STEM-EDX maps corresponding to Ni-K and Zn-K, respectively. Fig. 2d is an

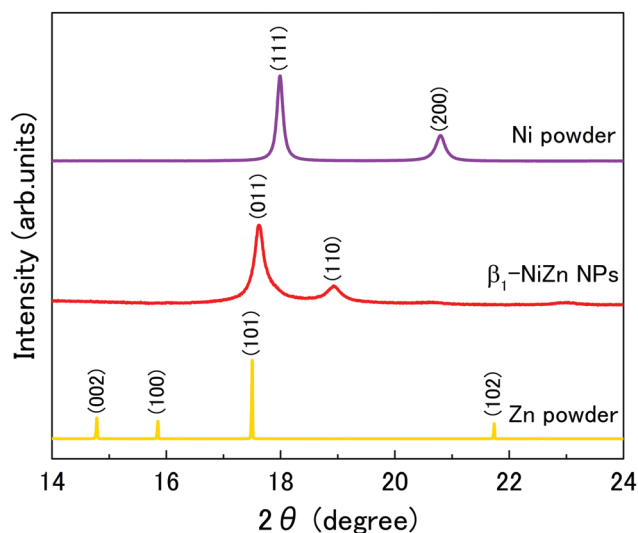


Fig. 1 Synchrotron powder XRD patterns of Ni powder, NiZn nanoparticles, and Zn powder at room temperature. The radiation wavelength was 0.636038 Å.

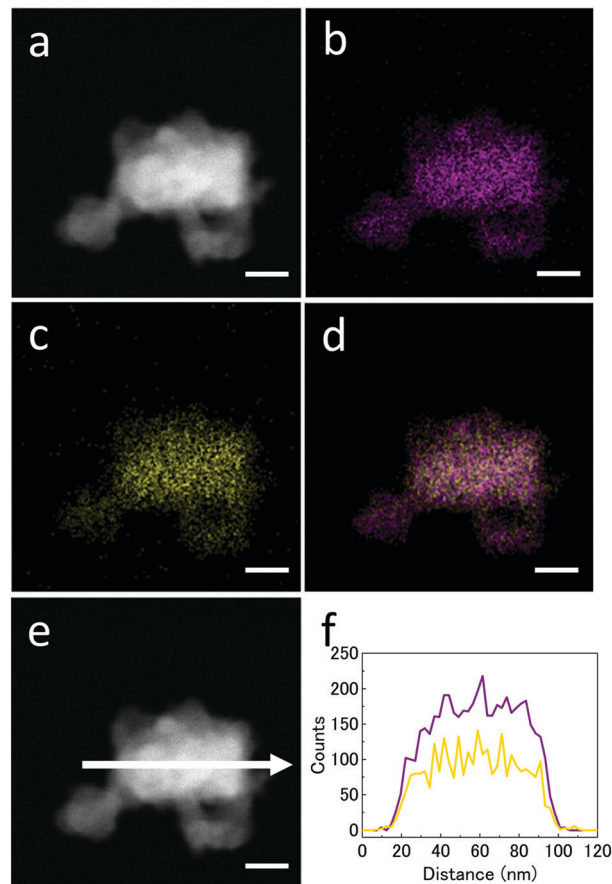


Fig. 2 (a) HAADF-STEM image, (b) Ni-K STEM-EDX map and (c) Zn-K STEM-EDX map of (a). (d) Overlay image of (b) and (c). (f) EDX line profiles across the NiZn nanoparticle along the arrow shown in (e). Ni and Zn are shown as purple and yellow lines, respectively. Scale bars correspond to 25 nm.

overlay map of the Ni and Zn chemical distributions. Fig. 2f shows EDX line scan profiles of the NiZn alloy nanoparticle shown in Fig. 2e. The line profiles confirmed that Ni and Zn were uniformly distributed over the whole nanoparticles. The average stoichiometry from the EDX data was Ni : Zn = 0.59 : 0.41. These results indicate that the synthesized nanoparticles are uniform NiZn alloy without any phase separation. We further investigated the detailed structure of the NiZn nanoparticles at atomic resolution. Fig. 3a and b show a high magnification HAADF-STEM image and line profiles of Ni and Zn for the NiZn nanoparticle recorded in the area along the black arrow shown in Fig. 3a, respectively. A regular arrangement of Ni and Zn atoms was clearly observed. Fig. 3d and e show STEM-EDX maps corresponding to Ni-K and Zn-K, respectively. Fig. 3c is an overlay map of Fig. 3d and e. The lattice image corresponds to the atomic arrangement of the β₁ phase of a NiZn intermetallic compound from the [010] zone axis, shown in Fig. 3a. These results indicate the successful synthesis of novel NiZn alloy nanoparticles where both Ni and Zn were uniformly mixed at the atomic level by a sodium-naphthalenide reduction method.

We also investigated the surface structure of the NiZn alloy nanoparticles. Fig. S3a and S3b (ESI[†]) show bright-field (BF)



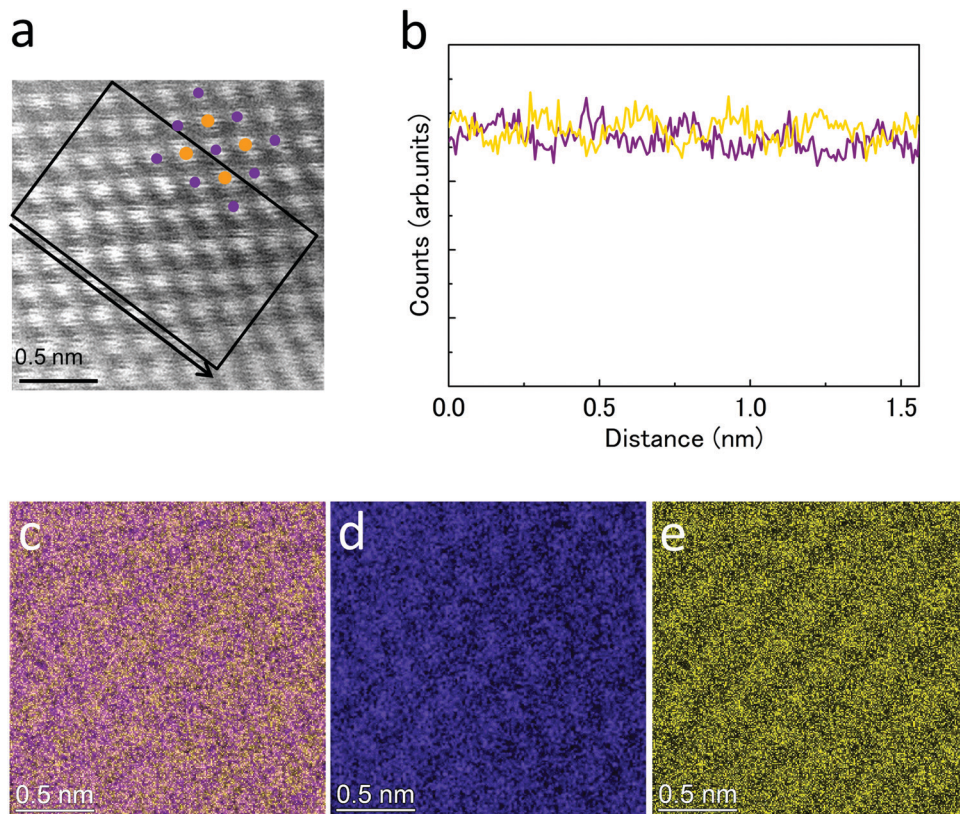


Fig. 3 (a) HAADF-STEM image and a model structure of β_1 phase of NiZn viewed from [010] zone axis (purple: Ni, yellow: Zn) (b) EDX line profiles of Ni (purple) and Zn (yellow) for a NiZn particle recorded in the area along the arrow shown in the STEM image (a). (c) Overlay image of (d) and (e). (d) Ni-K STEM-EDX map and (e) Zn-K STEM-EDX map of the NiZn nanoparticle.

STEM and HAADF-STEM images of a NiZn nanoparticle, respectively. A thin layer with a thickness of approximately 2 nm was observed at the surface of the nanoparticle. Fig. S4 (ESI[†]) shows the average stoichiometries determined from the EDX data. These results indicate that the thin layer consisted of NiZn oxide. In addition, the ratio of oxygen at the edge of the nanoparticle was greater than that at the center. Hence, oxidation occurred only at the surface after air exposure of the nanoparticles and the NiZn alloy nanoparticles were stable in air for more than half a year, although Zn is typically extremely easily oxidized. This result might be attributed to surface passivation by Ni.

The NiZn nanoparticles could be crystallized by sintering in a hydrogen atmosphere, although these were amorphous-like after the strong reduction reaction when the nanoparticle was an atomic-level mixture. When we adopted a different reduction process, phase-separated nanoparticles were obtained (Fig. S5 and S6, ESI[†]). In this case, the reduction rate of the Zn and Ni ions was not well controlled. Thus, we could not obtain NiZn alloy nanoparticles even if they were sintered under the same conditions. In other words, control of the reduction speed is the key to realizing NiZn alloy nanoparticles, which are mixed at the atomic level rather than the annealing process in a hydrogen atmosphere. This technique enables controlled simultaneous reduction of several materials and

might be applied to the synthesis of alloy nanoparticles of a variety of combination of elements.

In conclusion, we report the successful synthesis of NiZn homogeneous alloy nanoparticles that do not contain any noble metals by a sodium-naphthalenide reduction method. The synthesized nanoparticles were uniformly mixed at the atomic level and formed a β_1 phase NiZn intermetallic compound. These results indicate that Zn and Ni ions were concurrently reduced in a liquid phase through control of the reduction speed with a strong reduction agent. The crystal structure of the synthesized nanoparticles was different from those of their constituent elements. The alloy nanoparticles would exhibit new chemical and physical properties unlike those of the constituent Ni and Zn mono-metals. In addition, the NiZn nanoparticles were stable even in air, although Zn particles are typically easily oxidized. This synthetic technique based on a strong reduction agent might be applied to other alloy nanoparticles in a variety of combinations of elements, leading to the development of new materials with excellent properties.

Conflicts of interest

There are no conflicts to declare.



Acknowledgements

We thank Dr H. Kobayashi, Dr S. Kinoshita of Kyoto University for supporting the synthesis of NiZn nanoparticles. We are also thankful to Dr A. Sakai, Dr K. Nakajima, Dr S. Kusama, Dr H. K. Sato and Dr T. Saito of Panasonic Corporation for helping us to synthesize and characterize the nanoparticles.

Notes and references

- G. L. Brett, Q. He, C. Hammond, P. J. Miedziak, N. Dimitratos, M. Sankar, A. A. Herzing, M. Conte, J. A. Lopez-Sanchez, C. J. Kiely, D. W. Knight, S. H. Taylor and G. J. Hutchings, *Angew. Chem., Int. Ed.*, 2011, **50**, 10136–10139.
- X. Liu, A. Wang, X. Wang, C. Y. Mou and T. Zhang, *Chem. Commun.*, 2008, 3187–3189.
- W. Chen, J. Kim, S. Sun and S. Chen, *Langmuir*, 2007, **23**, 11303–11310.
- J. Zhang, D. N. Oko, S. Garbarino, R. Imbeault, M. Chaker, A. C. Tavares, D. Guay and D. Ma, *J. Phys. Chem. C*, 2012, **116**, 13413–13420.
- Z. Peng, B. Spliethoff, B. Tesche, T. Walther and K. Kleinermanns, *J. Phys. Chem. B*, 2006, **110**, 2549–2554.
- B. Zugic, L. Wang, C. Heine, D. N. Zakharov, B. A. J. Lechner, E. A. Stach, J. Biener, M. Salmeron, R. J. Madix and C. M. Friend, *Nat. Mater.*, 2017, **16**, 558–564.
- X. Huang, Z. Zhao, L. Cao, Y. Chen, E. Zhu, Z. Lin, M. Li, A. Yan, A. Zettl, Y. M. Wang, X. Duan, T. Mueller and Y. Huang, *Science*, 2015, **348**, 1230–1234.
- R. He, Y. C. Wang, X. Wang, Z. Wang, G. Liu, W. Zhou, L. Wen, Q. Li, X. Wang, X. Chen, J. Zeng and J. G. Hou, *Nat. Commun.*, 2014, **5**, 4327.
- C. Cui, L. Gan, M. Heggen, S. Rudi and P. Strasser, *Nat. Mater.*, 2013, **12**, 765–771.
- D. Alloyeau, C. Ricolleau, C. Mottet, T. Oikawa, C. Langlois, Y. Le Bouar, N. Braidy and A. Loiseau, *Nat. Mater.*, 2009, **8**, 940–946.
- S. Ma, M. Sadakiyo, M. Heima, R. Luo, R. T. Haasch, J. I. Gold, M. Yamauchi and P. J. A. Kenis, *J. Am. Chem. Soc.*, 2017, **139**, 47–50.
- P.-C. Chen, X. Liu, J. L. Hedrick, Z. Xie, S. Wang, Q.-Y. Lin, M. C. Hersam, V. P. Dravid and C. A. Mirkin, *Science*, 2016, **352**, 1565–1569.
- K. Kusada, M. Yamauchi, H. Kobayashi, H. Kitagawa and Y. Kubota, *J. Am. Chem. Soc.*, 2010, **132**, 15896–15898.
- K. Kusada, H. Kobayashi, R. Ikeda, Y. Kubota, M. Takata, S. Toh, T. Yamamoto, S. Matsumura, N. Sumi, K. Sato, K. Nagaoka and H. Kitagawa, *J. Am. Chem. Soc.*, 2014, **136**, 1864–1871.
- K. Kusada and H. Kitagawa, *Adv. Mater.*, 2016, **28**, 1129–1142.
- Q. Zhang, K. Kusada, D. Wu, T. Yamamoto, T. Toriyama, S. Matsumura, S. Kawaguchi, Y. Kubota and H. Kitagawa, *Nat. Commun.*, 2018, **9**, 510.
- S. Masuda, K. Shun, K. Mori, Y. Kuwahara and H. Yamashita, *Chem. Sci.*, 2020, **11**, 4194–4203.
- F. Wang, K. Kusada, D. Wu, T. Yamamoto, T. Toriyama, S. Matsumura, Y. Nanba, M. Koyama and H. Kitagawa, *Angew. Chem.*, 2018, **130**, 4595–4599.
- Q. Zhang, K. Kusada, D. Wu, N. Ogiwara, T. Yamamoto, T. Toriyama, S. Matsumura, S. Kawaguchi, Y. Kubota, T. Honma and H. Kitagawa, *Chem. Sci.*, 2019, **10**, 5133–5137.
- F. Studt, F. A. Pedersen, T. Bligaard, R. Z. Sørensen, C. H. Christensen and J. K. Nørskov, *Science*, 2008, **320**, 1320–1322.
- C. van der Wijst, X. Duan, I. S. Liland, J. C. Walmsley, J. Zhu, A. Wang, T. Zhang and D. Chen, *ChemCatChem*, 2015, **7**, 2991–2999.
- L. Jia, J. Yu, Y. Chen, P. Ning, Q. Guan, J. Gu, R. Miao and Q. Chen, *J. Supercrit. Fluids*, 2017, **126**, 79–88.
- S. Jana, J. W. Chang and R. M. Rioux, *Nano Lett.*, 2013, **13**, 3618–3625.
- H. Okamoto, *J. Phase Equilib.*, 2003, **24**, 280–281.
- C. Schöttle, P. Bockstaller, R. Popescu, D. Gerthsen and C. Feldmann, *Angew. Chem.*, 2015, **127**, 10004–10008.
- C. Schöttle and C. Feldmann, *Z. Anorg. Allg. Chem.*, 2016, **642**, 555–559.
- C. Schöttle, F. Gyger and C. Feldmann, *Chem. Ing. Tech.*, 2018, **90**, 427–435.
- S. Kawaguchi, M. Takemoto, K. Osaka, E. Nishibori, C. Moriyoshi, Y. Kubota, Y. Kuroiwa and K. Sugimoto, *Rev. Sci. Instrum.*, 2017, **88**, 085111.
- P. Nash and Y. Y. Pan, *Bull. Alloy Phase Diagrams*, 1987, **8**, 422–430.

