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Transport mechanisms in Co-doped ZnO (ZCO) and H-irradiated ZCO polycrystalline thin films

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In the present study, the electrical resistivity (ρ) as a function of the temperature (T) has been measured in polycrystalline ZnO, Co-doped ZnO (ZCO) and H irradiated ZCO (HZCO) samples, in the 300–20 K range. The achieved results show impressive effects of Co doping and H irradiation on the ZnO transport properties. The Co dopant increases the ZnO resistivity at high T (HT), whereas it has an opposite effect at low T (LT). H balances the Co effects by neutralizing the ρ increase at HT and strengthening its decrease at LT. A careful analysis of the ρ data permits to identify two different thermally activated processes as those governing the charge transport in the three materials at HT and LT, respectively. The occurrence of such processes has been fully explained in terms of a previously proposed model based on an acceptor impurity band, induced by the formation of Co-oxygen vacancy complexes, as well as known effects produced by H on the ZnO properties. The same analysis shows that both Co and H reduce the effects of grain boundaries on the transport processes. The high conductivity of HZCO in the whole T -range and its low noise level resulting from electric noise spectroscopy make this material a very interesting one for technological applications.

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1. Introduction

In Dilute Magnetic Semiconductors (DMS), a small amount of a magnetic element generates long range ferromagnetic ordering, generally *via* a carrier-mediated mechanism, that makes DMS important, potential components of spintronic devices.^{1–3} In the DMS family, Cobalt doped ZnO (ZCO) plays a primary role. In such a material, ZnO properties like large band gap (3.4 eV), long spin coherence length (10.7 nm at 10 K) and large exciton binding energy (60 meV), combine with the occurrence of ferromagnetism (FM) at room temperature (RT).^{4,5} Interestingly, the ZCO ferromagnetic properties can be improved by increasing the electron density through a co-doping with different ZnO shallow donors.^{6–12} Among possible shallow donors, atomic H stands out for some peculiar advantages. A mild incorporation of atomic hydrogen in the semiconductor lattice permits indeed an n-type doping of the material by performing post-growth treatments which minimize damaging effects.⁷

Moreover, in addition to the introduction of shallow donor levels and to the ensuing increase of conductivity,^{13–16} atomic H can neutralize electron traps at the surface of ZnO grain boundaries (GB)^{17,18} and it may reduce the O contents by favoring the formation of defects such as oxygen vacancies (V_O) and Zn interstitials (Zn-int). V_O have been proposed as components of Co- V_O complexes playing a key role in the occurrence of RT FM.^{19,20} Zn-int defects may further increase the n-type carrier density.

Many studies have been focused on the ZCO and H⁺-irradiated ZCO (HZCO) magnetic properties.^{6–12} However, in view of technological applications, it is also of crucial importance to understand the mechanisms governing charge transport processes in these materials. In this regard, preliminarily, it can be helpful to recall some previous results concerning the effects induced by Co doping and H irradiation on the ZnO properties. ZnO generally shows an n-type character due to shallow-donor native defects or unintentional dopants like, *e.g.*, Zn-int or Al atoms substituting the Zn ones, respectively. As mentioned above, H can significantly increase the ZnO and ZCO conductivity. In the latter material, however, it also gives rise, together with the Co dopants, to somewhat surprising phenomena.⁷ As a matter of fact, it has been observed that HZCO samples showing a larger ferromagnetic enhancement, directly related to the H dose, show at the same time a lower n-type carrier density, as if Co atoms had a sort of compensating acceptor-effect of the H donors.⁷ The same acceptor-like behavior is

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indicated by a reduction of the n-type carrier density observed when Co atoms are introduced in n-type ZnO.²⁰ Both results are quite surprising, a Co-induced compensation of shallow donors seems indeed incompatible with the fact that it is an iso-electronic impurity substituting a Zn atom. In the above cited studies,^{7,20} a model has been proposed to explain these puzzling findings, founded on the appearance of an acceptor impurity band close to the ZnO conduction band (CB), which originates from electronic states induced by Co-V_O complexes.¹⁹ This model, further supported by subsequent studies,^{12,21} will be recalled in the following discussion of present results.

As far as charge transport processes are concerned, many studies were focused on polycrystalline ZnO.^{22–31} Only few studies addressed the same processes in ZCO^{27,32} with only two papers devoted to transport mechanisms in hydrogenated ZnO.^{18,29} No papers have been published on transport processes in HZCO, at the best of the author knowledge. Such a lack of information and the previous results showing that both Co doping and ZCO H co-doping can affect the ZnO conductivity, have motivated the present study of transport properties in polycrystalline PLD (pulsed laser deposition) grown ZnO, ZCO and HZCO thin films, where the electrical resistivity has been investigated from RT down to 20 K.

The achieved results were beyond our expectations by showing that both Co doping and H co-doping produce impressive effects on the ZnO resistivity. More specifically, it is shown that Co doping induces unexpected, opposite effects on the ZnO resistivity in the high temperature (HT) and low temperature (LT) regimes, respectively: it increases the ZnO resistivity in the former case while it induces the reverse in the latter one. On the other hand, H neutralizes the Co-induced increase of resistivity at HT whereas it enhances the resistivity decrease at LT, by realizing a sort of balancing effect, resulting in an overall reduction of the ZnO resistivity in the whole investigated temperature range.

A careful analysis of our results has allowed us to identify two different thermally activated processes playing a primary role in the transport mechanisms at the HT and LT regimes, respectively. The occurrence of such processes has been fully explained on the base of the mentioned model explaining the apparent Co acceptor behavior and on previously observed effects induced by Co doping and H irradiation on the ZnO properties.^{7,20}

The above investigations have been integrated with electric noise spectroscopy measurements of the HZCO sample showing that this material combines the known FM properties, the low resistivity and good mobility with a quite low noise level, thus making this material a very interesting candidate for technological applications.

2. Experimental section

ZnO and 5% Co-doped ZnO films (ZCO), were deposited by pulsed laser deposition using a Nd:YAG laser, operating at $\lambda = 355$ nm, with a fluence of 2 J cm^{-2} and a repetition rate of 10 Hz, and Zn_{0.95}Co_{0.05}O pellets as target. These films were

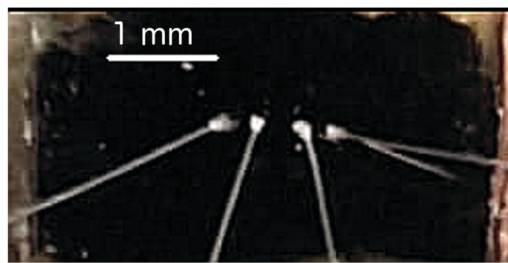


Fig. 1 Micro photograph of the sample with the inline bonding pads. Square 150 nm thick Al pads have 200 μm sides and are separated by 200 μm wide gaps. The voltage pads distance is $l = 200 \mu\text{m}$. The sample width is 5 mm.

grown on optically transparent Al₂O₃(001) single crystal substrates, placed at a distance of 5 cm from the target, at 500 °C and 2×10^{-5} mbar O₂ pressure with a growth rate of about 4 Å s^{-1} .^{7,12,20,21,33–35} The ZCO films were irradiated at 400 °C with a H⁺ dose of 4.5×10^{18} ions per cm² produced by a Kaufman source. Structural and microstructural characterizations, ruling out the presence of noticeable Co-related secondary phases, are reported elsewhere;^{7,12,20} the thickness, determined by scanning and transmission electron microscope measurements, was $(300 \pm 30) \text{ nm}$.¹² Carrier concentration (n) and mobility (μ) of the ZnO, ZCO and HZCO samples were estimated at RT by Hall measurements.^{7,20}

Four-probe dc resistance measurements were carried out by a standard technique with an inline geometry for the contacts, ultrasonically bonded to Al pads defined by optical lithography and lift-off, see Fig. 1. The samples were biased with a Keithley mod. 220 current source and the output voltage was amplified with homemade low-noise electronics and recorded with a Keithley mod. 2002 voltmeter. Resistivity was derived from the R vs. T measurements taking into account film thickness and geometrical factors.

A similar configuration of the contact pads was used for temperature and voltage-noise characterizations, made with a closed-cycle refrigerator system (Janis Research) operating in the 8–325 K range with an active temperature stabilization of 0.2 K. The data acquisition was performed using bias and readout electronics optimized for low-noise measurements,^{36,37} whose effectiveness was already proven in several condensed matter systems.^{38–40}

3. Results and discussion

3.1 Electric transport processes

The measured resistivity (ρ) of ZnO, ZCO and HZCO samples is shown in Fig. 2 as a function of the temperature, T . The curves in the figure exhibit an increase in resistivity with decreasing T , according to a general semiconducting behavior. Most important, the ZnO and ZCO curves show a striking effect of Co doping on the ZnO resistivity.

As mentioned above, Co induces indeed opposite effects on the ZnO resistivity by increasing its value in the HT regime and producing the opposite effect in the LT one, respectively.



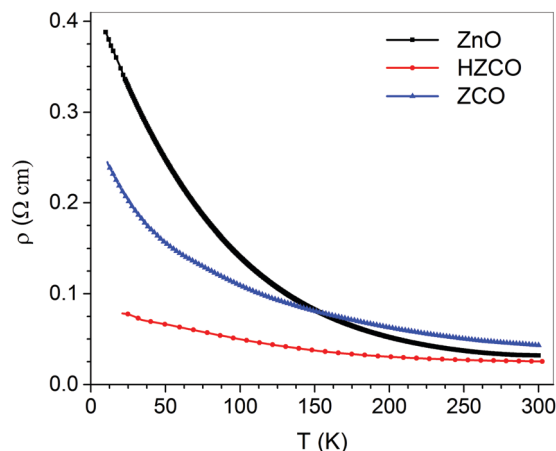


Fig. 2 Resistivity (ρ) measurements as a function of temperature for ZnO, ZCO, and HZCO films in the range 20–300 K.

This produces an unexpected crossing of the two curves at an intermediate temperature. Even H has remarkable effects by neutralizing the ρ increase induced by Co doping at HT and strengthening the ρ reduction at LT. Moreover, it also induces the lowest ρ values in the whole sampled T -range, thus avoiding any crossing among the ρ curves.

The shapes of the ρ curves suggest the occurrence of different thermally activated processes.²⁶ In order to get a first information on such processes, our ρ data were fitted in terms of ρ^{-1} conductivity in the whole temperature range to the following equation:^{41,42}

$$\rho^{-1} = pe^{-E_1/kT} + qe^{-E_2/kT} + me^{-E_3/kT} \quad (1)$$

where p , q , and m are temperature-independent conductivity parameters and E_i are thermal activation energies describing, in principle, the electronic conduction in high (E_1), intermediate (E_2), and low (E_3) temperature regimes. The results of the fitting procedure are shown in Fig. 3 and summarized in Table 1.

It turns out that the materials show a different number of thermally activated processes in the investigated T -range. In ZnO, two processes are active in the 300–170 K and 170–20 K T ranges, respectively. Three processes occur in ZCO in the 300–240 K, 240–154 K and 154–20 K ranges, respectively, whereas two processes occur in HZCO in the 300–117 K and 117–20 K ranges, respectively. Thus, in order to perform a homogeneous comparison of the three materials, we specify the 300–240 K and the 117–20 K ranges as corresponding to the HT and LT regimes, respectively.

Firstly, the results in Table 1 confirm the significant effect of Co doping on the ZnO conductivity mechanisms, as shown by the appearance of a third thermally activated process, as well as the balancing effect of H that restores the number of thermally activated processes back to two. Then, to identify the kind of thermally activated process which dominates the conductivity mechanisms in the three materials, we have analyzed the above results by distinguishing the HT regime from the LT one. Such an analysis has been performed by taking into account the polycrystalline nature of our samples,^{7,20} where grain boundaries

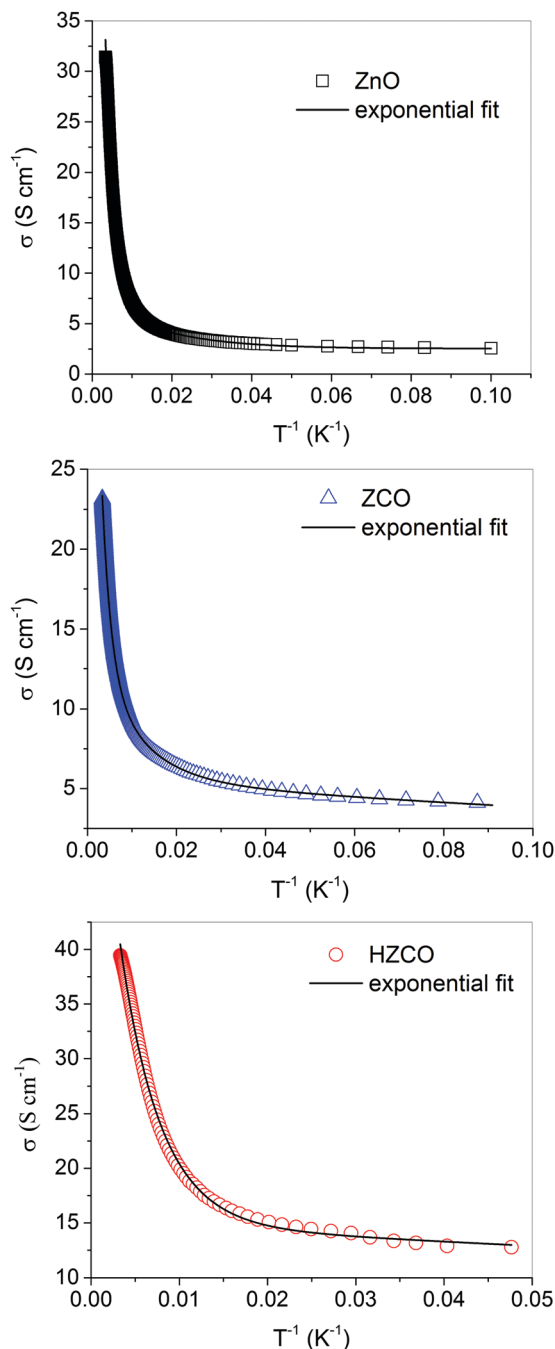


Fig. 3 Conductivity (ρ^{-1}) as a function of T^{-1} for ZnO, ZCO, and HZCO films in the range 20–300 K; solid lines represent a fit to eqn (1) (see text).

(GB) could play a significant role in the transport processes. In particular, in the case of ZnO, its surface reactivity and n-type nature, which implies a noticeable presence of electrons in the CB, induce the formation of O_2 -derived, negatively charged defects at the grain boundary surfaces. Such defects produce a depletion layer of electrons close to the surface and a buildup of negative charge at grain boundaries. This phenomenon gives rise to a band bending in the bulk region near the GB inducing a potential barrier between the grains (V_b) that can inhibit inter-grain electronic transport. By the way, note that the V_b 's can be of



Table 1 Activation energies derived from data fitting to eqn (1) (see the text) are reported in the table together with 300 K and 20 K ρ values, RT carrier concentration (n), and RT mobility (μ) values^{7,20}

Samples	E_1 (K) (meV)	E_2 (K) (meV)	E_3 (K) (meV)	n ($\times 10^{18}$ cm ⁻³)	μ (cm ² V ⁻¹ s ⁻¹)	ρ @300 K (Ω cm)	ρ @20 K (Ω cm)
ZnO	417 $\pm$ 5	—	51 $\pm$ 5	9.0 $\pm$ 0.1	7 $\pm$ 1	0.032 $\pm$ 0.001	0.350 $\pm$ 0.001
	36.0 $\pm$ 0.4	—	4.6 $\pm$ 0.4				
ZCO	492 $\pm$ 4	105 $\pm$ 3	3.9 $\pm$ 0.2	2.0 $\pm$ 0.1	15 $\pm$ 1	0.043 $\pm$ 0.001	0.220 $\pm$ 0.001
	42 $\pm$ 2	9.0 $\pm$ 0.3	0.34 $\pm$ 0.01				
HZCO	220 $\pm$ 5	—	3 $\pm$ 1	3.4 $\pm$ 0.1	21 $\pm$ 1	0.025 $\pm$ 0.001	0.080 $\pm$ 0.001
	19.0 $\pm$ 0.4	—	0.25 $\pm$ 0.09				

variable height possibly depending on the surface microstructure (e.g., on the O₂-like defect density).

In principle, in polycrystalline materials, the conduction mechanisms can be affected by thermally activated processes which control the intra-grain electron transport (ig processes) or the inter-grain one through the GB barriers (gb processes). In a given T -range, the ig or gb process requiring the largest activation energy is expected to be the “bottle-neck” process that governs the conduction mechanisms.

3.1.1 High temperature regime. Based on the above considerations, we first analyze the HT regime. Three different thermally activated processes corresponding to the E_1 activation energies of Table 1 can be taken into account in an n-type semiconductor, like, ZnO. The first one refers to an intra-grain conduction process which arises from mobile electrons in the CB, thermally excited to this band from shallow donor states (D). The other two processes refer instead to the inter-grain charge transport, involving the passing of the V_b barrier at the grain boundaries. In principle, the current can flow between grains by thermionic emission (TE) over the potential barrier or, alternatively, by thermally-assisted quantum-mechanical tunneling (TAT) of electrons through the V_b barriers. An indication on the main charge transport mechanism occurring in the present case can be obtained by applying a simple criterion based on the so-called E_{00} tunneling parameter:⁴³

$$E_{00} = 18.5 \times 10^{-12} \left(\frac{n}{\varepsilon m^*} \right)^{\frac{1}{2}} \text{eV} \quad (2)$$

where n is the carrier density, ε the relative dielectric constant, and m^* the reduced electron effective mass of a given semiconductor.

If $E_{00} < kT$ condition is satisfied a TE process is the relevant charge transport mechanism. In the case of our samples, where $n_{\text{ZnO}} = 9 \times 10^{18} \text{ cm}^{-3}$, $n_{\text{ZCO}} = 2 \times 10^{18} \text{ cm}^{-3}$, $n_{\text{HZCO}} = 3.4 \times 10^{18} \text{ cm}^{-3}$, $m^* = 0.24$, and $\varepsilon(\text{ZnO}) = 8.5$,²⁵ the E_{00} value is larger than kT for ZnO (~ 41 meV) and comparable to kT for ZCO and HZCO (24 meV) in the whole 300–240 K range.⁴⁴ Thus, thermionic emission is not the prevalent process at HT.

The occurrence of TAT processes can be ruled out as well by comparing the ZnO E_1 value with the ZCO one and by taking into account the results of a previous KPFM (Kelvin Probe Force Microscopy) study on the effects produced by Co doping on the ZnO surface reactivity and electronic properties.²¹ Such results show that Co introduction strongly reduces the absorption of negative O₂-derived species at the ZnO surface, thus inducing a strong reduction of the V_b barriers. We anticipate that such a

Co effect is fully confirmed by the results achieved here at the LT regime, that will be discussed in the following paragraphs. Now, if a TAT process dominated the ZnO conduction mechanisms, a reduction of the V_b height induced by the Co doping would give rise to a reduction of the E_1 value on going from the ZnO to the ZCO, contrary to the results of Table 1. This rules TAT processes out in the ZnO and ZCO cases. The same conclusion can be drawn in HZCO. In fact, the above findings imply a minor role of grain boundaries in the ZCO transport processes at HT. The same must hold for HZCO since H passivates the O₂-like negative surface defects,^{17,18} thus further reducing the V_b height. We can conclude that the thermal excitations from the D levels to the CB dominate the transport processes at the HT regime in all of the three materials.

The occurrence of this kind of thermally activated process and the model based on an acceptor-like impurity band induced by Co-V_O complexes mentioned above perfectly explain the increase of the E_1 values at HT in Table 1 and that of the corresponding ρ values in Fig. 2 when going from ZnO to ZCO. The Co-V_O impurity band can subtract indeed electrons from the shallow donor levels, thus producing an increase of the activation energy required for electron excitation to the CB and of the ρ values as well.

The introduction of atomic H in the ZCO sample induces the formation of shallow donor levels which compensate for the effects of the acceptor Co-V_O impurity band.^{7,12} This accounts for a decrease of the activation energy and resistivity values, which result to be both smaller even than the corresponding ZnO values. Thus, atomic H fully neutralizes the effects induced by Co doping on the ZnO transport processes at HT.

3.1.2 Low and intermediate temperature regimes. In the case of the LT regime, we can hypothesize two kinds of thermally activated processes requiring small activation energies: (i) intra-grain conduction processes controlled by the “hopping” of electrons between localized impurity states, which may occur in the Mott or Efros VRH (Variable Range Hopping) regimes^{45,46} and (ii) inter-grain TAT processes. First, we analyzed our resistivity data in the framework of the Mott VRH model,⁴⁵ which predicts a ρ temperature dependence according to the $\rho = \rho_0 \exp(T_M/T)^{1/4}$ law, in the range $T < 60$ K. Fitted values of the characteristic Mott temperature T_M are 2400 K, 3164 K, and 2687 K, for the HZCO, ZCO, and ZnO samples, respectively. Two conditions have to be fulfilled for the Mott VRH model to be valid:⁴⁷

(a) the electrons must hop a mean distance R considerably greater than the localization length ξ ,⁴⁸ i.e. $R/\xi > 1$.



(b) the mean hopping energy difference between sites (W) has to be larger than kT , i.e. $W/kT > 1$.

We calculated such quantities following the Mott relations:

$$\frac{R}{\xi} = \frac{3}{8} \left(\frac{T_M}{T} \right)^{\frac{1}{4}} \quad (3)$$

$$\frac{W}{kT} = \frac{1}{4} \left(\frac{T_M}{T} \right)^{\frac{1}{4}} \quad (4)$$

Although (a) is satisfied for $T < 40$ K, (b) is not fulfilled in the same T -range, indicating that Mott model is not applicable to our data. Furthermore, a description of our data in the framework of the Efros–Shklovskii VRH model ($\rho = \rho_0 \exp(T_{ES}/T)^{1/2}$), with a fitted characteristic temperature $T_{ES} \sim 100$ K, shows that the conditions for the model validity ($R/\xi = (1/4)(T_{ES}/T)^{1/2} > 1$ and $W/kT = (1/2)(T_{ES}/T)^{1/2} > 1$) are once again not fulfilled.⁴⁷

Therefore, TAT processes dominate the conductivity mechanisms at LT, thus implying a significant GB role. The E_3 activation energies estimated indicate that V_b barriers of small amplitude oppose to the charge transport at LT in all the three materials. As anticipated, a ZCO E_3 value significantly smaller than the ZnO E_3 one fully supports a reduction of the V_b barriers induced by the Co doping. The ZCO E_3 activation energy shows a further slight decrease in the case of HZCO. The resulting, overall activation energy trend fully agrees with that of the corresponding ρ values of Fig. 2 at LT. The activation energy and ρ decrease following the Co introduction in ZnO is explained very well by the inhibiting effect of such a doping on the formation of O_2 -derived defects at the ZnO GB surfaces. On the other hand, the known passivation effects of the same surface defects induced by atomic H^{17,18} fully account for the further decrease of the HZCO E_3 activation energy with respect to the ZCO E_3 one. As mentioned above, atomic H introduces also shallow donor levels in the ZnO band structure, which can contribute to a further reduction of ρ values, thus enhancing the effects of Co doping on the ZnO conduction processes at LT.

Altogether, the above considerations give a consistent explanation of the Co doping and H co-doping effects on the ZnO transport properties. Now, it is worth to reconsider the third thermally activated process appearing at an intermediate T for ZCO only. Even in this case, Mott or Efros VRH mechanisms can be ruled out as an origin for that third process. Moreover, we can also rule out processes related to the thermal excitation of electrons from D levels because Co doping cannot induce the formation of donor levels more shallow than those introduced by unintentional dopants or atomic H, as it is indicated by a comparison of the E_2 ZCO value with the E_1 ZnO and HZCO values in Table 1. Thus, our suspects turn to TAT processes. The occurrence of such a kind of processes seems reasonable since it could be related to a variability of the V_b barriers in ZCO with respect to ZnO and HZCO. In fact, we can assume that at LT these barriers have similar, small heights in ZnO. In ZCO, instead, a variability of the Co concentration in the grains could induce a V_b variability responsible of the appearance of slightly higher barriers which, in turn, would be at the origin of

the process requiring slightly higher activation energies (9.0 meV vs. 4.6 meV) at intermediate T . The action of atomic H, given its high diffusion capability, could produce a homogeneous reduction of the ZnO V_b heights, thus restoring the occurrence of only two thermally activated processes at HT and LT, respectively.

We conclude the present section with two final comments. First, the above discussion of LT results indicates a progressive reduction of the GB V_b 's on going from ZnO to ZCO, and to HZCO. The same reduction occurs at HT where it is reflected by the RT μ values that significantly increase from ZnO to ZCO and further increase to HZCO, see Table 1. The same table shows that n values lower from ZnO to ZCO. This may indicate that the μ increase due to a reduction of the V_b effects is possibly accompanied by a reduction of electron scattering. On the other hand, n values increase from ZCO and HZCO. In this case, therefore, the μ increase can be exclusively attributed to a V_b reduction. Second, as far as the investigated parameters of Table 1 depend on the Co content, measurements of the carrier density give values of $7.6 \times 10^{18} \text{ cm}^{-3}$, $3.6 \times 10^{18} \text{ cm}^{-3}$, and $2.0 \times 10^{18} \text{ cm}^{-3}$ for ZCO samples 2%, 4%, and 6% Co-doped, respectively. These values, together with the $9.0 \times 10^{18} \text{ cm}^{-3}$ n value measured for the ZnO sample, indicate a noticeable decrease of the carrier density for a Co doping level above 4%. This value can be considered, therefore, as a lower bound for Co concentrations able to produce an effective action of the acceptor impurity band. Given the role of such a band in both the HT and LT transport processes, we estimate that the same lower bound similarly affects the μ , ρ , and activation energy values of Table 1. For Co concentrations larger than 6%, the occurrence of a further reduction of the carrier density depends on the Co- V_O complex concentration, more specifically on the V_O content whose increase may require post-growth treatments.

3.2 Electric noise spectroscopy

In the light of the above results, HZCO can be considered a very interesting material in view of possible opto-spintronics applications due to the coexistence of improved RT magnetic/magneto-optic properties with n-type doping and reduced GB effects and, hence, large electric conductivity and mobility. It is, therefore, useful to investigate its charge carriers fluctuation mechanism, by means of the well-known electric noise spectroscopy. Such technique is based on the study of the voltage fluctuations in the frequency domain, represented by the voltage-spectral density S_V , and their dependence on externally controlled parameters such as temperature, bias current, and air pressure.⁴⁹ Measured S_V values of HZCO are shown in Fig. 4a. The measurements were performed in vacuum at RT and in a bias current range varying from 0 to 2 mA and are characterized by a $1/f$ -type spectrum. At high frequencies, the $1/f$ noise becomes negligible and S_V is dominated by thermal noise, having a white spectrum. The behavior shown in Fig. 4a is expected in the case of films with oriented crystalline structure, as already reported for ZnO samples deposited on glass and Pt/Si substrates,⁵⁰ for Al-doped ZnO thin films,^{51,52} and for ZnO-based varistors.⁵³



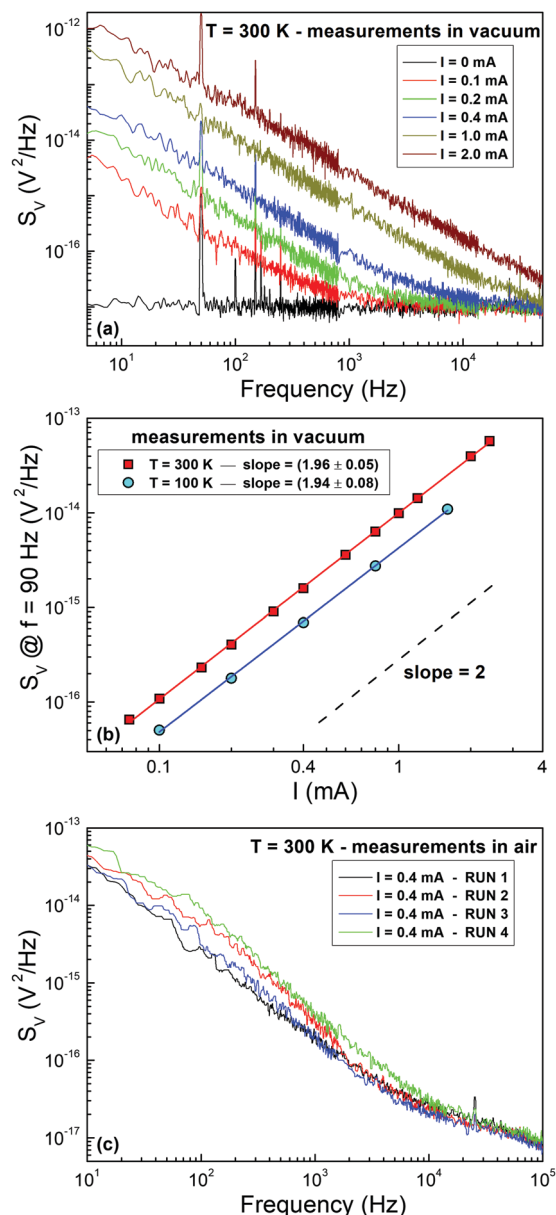


Fig. 4 (a) S_V at 300 K for different dc bias current values measured in vacuum. (b) Bias current dependence, measured in vacuum, of the $1/f$ noise amplitude at 300 and 100 K, showing a quadratic behavior. (c) Measurements of S_V at 300 K and at 0.4 mA, in air.

The study of the bias current dependence of the $1/f$ amplitude has also been performed, since it is a valuable method capable for extracting useful information about the electrical transport mechanisms at work, as extensively reported for a large variety of materials, such as magnetic compounds,^{54–57} superconducting thin films and devices,^{58,59} and carbon nanotube composites.^{60–62} Fig. 4b shows a quadratic dependence of the $1/f$ noise as a function of the bias current, both in the HT (300 K, red squares in Fig. 4b) and in the LT region (100 K, blue dots in Fig. 4b). This behavior, and the decrease of the noise level on lowering the temperature (also evident in Fig. 4b), is typically expected when the noise source is due to resistance

fluctuation, according to the Dutta and Horn model,^{63,64} where the electric transport is explained in terms of thermally activated processes. This result further supports the conduction mechanisms here proposed to explain electric transport in HZCO.

In view of possible applications of HZCO in opto-spintronic devices, noise measurements have been also performed in air and at RT. The S_V curves, shown in Fig. 4c, are acquired at the same bias current value of 0.4 mA, and are characterized by a $1/f$ -type spectrum superimposed to a single small amplitude Lorentzian noise component, having a typical a $1/f^2$ dependence. We note that the cutoff visible at the highest frequencies is due to finite bandwidth of the electronic readout. The behavior in air differs significantly from the one observed for metal oxide gas sensors,^{65,66} where high amplitude multi-Lorentzian components in the noise spectra are ascribed to the effect of the adsorbed molecule's density fluctuation on the GB barrier height.⁶⁷ This is largely different from the noise spectra observed in our case in agreement with the positive role played by H irradiation in decreasing the effect of grain boundaries in polycrystalline Co-doped ZnO thin films.

Moreover, additional information can be obtained from the evaluation of the intrinsic noise level of HZCO, by using the Hooge empirical relation.⁴⁹ The quantity, $[\alpha \cdot \mu]_{\text{eff}}$, where α is the Hooge parameter and μ the mobility, is related to the $1/f$ noise amplitude for unit surface by the equation:^{49,51}

$$[\alpha \cdot \mu]_{\text{eff}} = \frac{fSL^2}{V^2 q R_{\text{sh}}} \quad (5)$$

where $L = 200$ μm is the voltage pads distance, q is the electron charge, and R_{sh} is the sheet resistance. At RT, the normalized voltage-spectral density $f \cdot S_V/V^2$ is extracted from the experimental data shown in Fig. 4b and has the value of $(4.8 \pm 0.1) \times 10^{-11}$. Substituting in eqn (5), it is straightforward to compute $[\alpha \cdot \mu]_{\text{eff}} \approx 122 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. This value has to be considered as a close and reliable estimate of the real value, even though the investigated sample is an unpatterned film. A comparison with noise results reported in literature shows that the noise level of HZCO is very close to the one reported for very high-quality and less-noisy ZnO films deposited on glass substrates,⁵⁰ and in Al-doped ZnO deposited at RT.⁵² Furthermore, the HZCO noise level is found up two orders of magnitude lower than that reported for ZnO films deposited on Pt substrates,⁵⁰ for Al-doped ZnO films deposited in the temperature range (200–400) $^\circ\text{C}$,⁵¹ and for ZnO-based varistor structures.⁵³ The above experimental results indicate therefore the HZCO material as a good candidate for technological applications.

4. Summary and conclusions

In the present study, the effects of Co doping and of a subsequent H irradiation on the conductivity mechanisms of polycrystalline ZnO thin films have been investigated by measuring the resistivity of ZnO, ZCO and HZCO samples as a function of the temperature in the RT–20 K range. This investigation has produced three major results.



First, Co doping has a deep impact on the ZnO transport processes by inducing a resistivity increase at HT, whereas the opposite occurs at low T . This results in a striking crossing of the ZnO and ZCO ρ curves at an intermediate T . Second, the H irradiation rebalances the Co effects by neutralizing the ρ increase at high T and strengthening the ρ decrease at low T . Thus, the combined effect of H and Co doping gives rise to an overall decrease of the ZnO resistivity in the whole sampled T -range.

Third, a careful analysis of the present ρ data has fully clarified the mechanisms governing the charge transport in the three materials by identifying two different thermally activated processes playing a dominant role at HT and LT, respectively. In detail, a thermally activated electron excitation from shallow donor levels to the CB dominates the conduction mechanisms in the three materials in the former regime. On the other hand, a thermally assisted tunneling process governs the charge transport at LT. Grain boundary potential V_b barriers play a primary role in the charge transport only in the LT regime. Remarkably, their effects are reduced by the Co doping with a further reduction by H irradiation of ZCO.

Present results are fully explained in terms of a previously proposed model based on an acceptor impurity band, induced by the formation of Co- V_O complexes, as well as reported effects of Co doping²¹ and H irradiation on the ZnO GB potential V_b barriers.^{17,18}

Finally, electric noise spectroscopy measurements of the HZCO sample, performed in vacuum and air, support the occurrence of thermally activated processes at HT as well as a minor role of GB V_b effects. They also show that the HZCO noise level is significantly lower than that reported in previous papers for ZnO films. Thus, the high conductivity found for HZCO, both in HT and LT regimes, together with its enhanced magneto-optic properties and the excellent noise characteristic, make this material a very interesting candidate for opto-spintronic applications.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

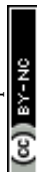
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