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Effect of temperature on the transformation of amorphous calcium magnesium carbonate with near-dolomite stoichiometry into high Mg-calcite†

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High Mg-calcite (HMC) is thermodynamically unstable under ambient conditions, yet it has been found in many biogenic and abiotic depositional settings with Mg contents up to 50 mol%. The elevated Mg content of HMC was frequently attributed to an amorphous calcium magnesium carbonate (ACMC) precursor, but the effects of transformation conditions on the Mg content of HMC are still debated and far from being quantified. Therefore, the transformation of ACMC with near-dolomite stoichiometry (~48 mol% Mg) into HMC has been studied in a MgCl_2 - NaHCO_3 buffered solution at moderate pH (7.6) and temperatures from 10 to 80 °C. The obtained chemical data show that the apparent solubility of ACMC (K_{ACMC}) is lower at higher temperature and the relative increase of K_{ACMC} as a function of the Mg content is similar between 10 and 80 °C which can be assessed by the equation: $\log(K_{\text{ACMC}}) = 0.01629 \cdot [\text{Mg}]_{\text{ACMC}} - 0.0001096 \cdot T^2 + 0.0545 \cdot T - 12.919$, where T is reported in Kelvin and $[\text{Mg}]_{\text{ACMC}}$ refers to the amount of Mg in ACMC in mol%. The Mg content of the final HMC increases from 5 to 40 mol% with increasing temperature, thus is significantly lower compared to the Mg content of the ACMC precursor. Our findings argue for a dissolution and re-precipitation process during ACMC transformation, where at higher temperatures the Mg incorporation into HMC is enhanced due to (i) the reduced solvation energy barrier of aqueous Mg^{2+} and (ii) the high prevailing molar $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution after its reaction with the ACMC. Notably, the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the latter solution is controlled by the temperature-dependent solubility of ACMC. Finally, the findings indicate that the nanocrystalline HMCs formed *via* transformation of ACMC are physically defective from rapid crystal growth and have higher solubilities compared to well-crystallized Mg-bearing calcites.

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1. Introduction

Calcite is the most abundant calcium carbonate (CaCO_3) mineral in marine and terrestrial environments.^{1,2} In the crystal structure of calcite, Ca^{2+} can be substituted by other

divalent cations present in natural waters. Predominant in these replacement reactions is the substitution of Ca^{2+} by Mg^{2+} .^{3–5} In the literature, calcite containing >4 mol% Mg is usually referred to as high Mg-calcite (HMC; $\text{Ca}_{1-x}\text{Mg}_x\text{CO}_3$, $x > 0.04$), whereas calcite with lower Mg content is referred to as low Mg-calcite (LMC; $\text{Ca}_{1-x}\text{Mg}_x\text{CO}_3$, $x < 0.04$).^{5,6} Abiogenic HMCs with <20 mol% Mg form marine cements consolidating sedimentary materials, *e.g.* in shallow tropical carbonate factories,³ in zones affected by anaerobic oxidation of methane⁷ and in coastal sabkha settings.⁸ Biogenic HMCs with up to 25 mol% Mg are a common building material of *e.g.* skeletons of coralline algae, foraminifers and echinoderms.^{3,9–12} The Mg content can be even higher (40–45 mol%) in structural elements of sea urchin teeth.^{13,14} A few examples of HMC with up to nearly 50 mol% Mg have been reported in natural depositional settings which are usually rich in microbial activity such as in algal sediments from a hypersaline lake (salt pond) of San Salvador Island, Bahamas,¹⁵ in microbial mats of the Lagoa Vermelha,

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† Electronic supplementary information (ESI) available: Table S1: ICSD database code and reference used for mineral phase quantification; Table S2: concentrations of free Ca^{2+} and Mg^{2+} ions and calculated $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratios of the experimental solutions; Table S3: activities of Ca^{2+} , Mg^{2+} and CO_3^{2-} ions in the experimental solutions and calculated solubility product values (K) for ACMC and HMC; Table S4: summary of solubility data of amorphous Ca-(Mg)-carbonates; Fig. S1: XRD pattern, SEM image and thermal analysis of synthesized ACMC; Fig. S2: temporal evolution of pH and HCl solution addition to the experimental solution; Fig. S3: XRD patterns of solids collected during ACMC transformation experiments at a 2θ range from 4° to 65°; Fig. S4: XRD pattern and SEM image of the reference Coorong dolomite. See DOI: 10.1039/d0ce01679a


Brazil,¹⁶ in microbial films of the Seroe Domi Formation of Curacao, The Netherlands¹⁷ and in sediments of the shallow, alkaline Lake Neusiedl, Austria.¹⁸ Note here that HMC phases with near-dolomite stoichiometry, but lacking cation ordering, are also referred to as very high Mg-calcite (VHMC) or protodolomite.⁵

Despite several decades of research, the formation of HMC and VHMC in natural systems is a poorly understood process because (i) the (V)HMC structure is destabilized by the uptake of Mg^{2+} (ref. 19) and (ii) the inorganic precipitation of HMC with >20 mol% Mg in the laboratory under ambient conditions remains an enigma as high aqueous $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratios inhibit the nucleation and growth of HMC from solution and favour the precipitation of aragonite (CaCO_3).^{20–23} The inhibiting effect of aqueous Mg^{2+} on the formation of calcite originates from the strong affinity of the Mg^{2+} ion for the aqueous solution which is reflected by its high hydration Gibbs free energy²⁴ and low rate of exchange of water molecules in its hydration sphere.^{25,26} The dehydration of the hydrophilic Mg^{2+} ion at the lattice growth site is therefore one of the most rate-limiting steps during the formation of (V)HMC.

Experimental studies on mineral formation, however, have shown that besides direct precipitation from solution (classical formation pathway), carbonate minerals can be formed *via* the transformation of amorphous precursor phases.^{27,28} Evidence for this alternative formation pathway also comes from calcifying organisms, where calcite formation often proceeds through an amorphous calcium carbonate (ACC) precursor phase.^{29,30} In this context, it has been suggested that Mg-containing ACC is a necessary prerequisite for the formation of HMC¹⁰ and VHMC.³¹ In most studies, the formation of amorphous calcium magnesium carbonate (ACMC) was induced by mixing a Na_2CO_3 solution (pH > 11) with (Ca, Mg) Cl_2 solutions (pH = 6–7) at room temperature.^{10,31–35} The chemical composition of the initial solution for ACMC synthesis (*e.g.* Mg/Ca ratio, pH) has been suggested to control the amount of Mg incorporated into the ACMC and the final HMC.^{36–38} Specifically, the formation of VHMC *via* ACMC has been achieved from synthesis in aqueous solutions with high Mg/Ca ratios at high pH (~ 10), high carbonate-to-calcium ratios (>2:1) or by thermal treatment (>40 °C).^{31,36,38–40} However, in most studies, the chemical compositions of the reactive solutions and solids during the formation of ACMC and its transformation into Mg-calcite were not concerned or even not documented. In particular, the effect of temperature on the Mg content of HMC formed *via* the amorphous pathway in aqueous solution has not been investigated in earlier studies. In order to better understand the fate of Mg during the transformation of ACMC into HMC, we investigated the transformation behaviour of synthetic ACMC with near-dolomite stoichiometry (48 mol% Mg) in solution at moderate pH (7.6) and reaction temperatures ranging from 10 to 80 ± 1 °C. The aim of the present study is to provide a more detailed understanding of the interplay between the

chemical composition of (i) ACMC, (ii) the corresponding reactive solution and (iii) (V)HMC formed *via* ACMC in the apparent aqueous environment. Moreover, the obtained chemical data were used to assess the solubility of ACMC and HMC.

2. Methods

2.1 Synthesis of amorphous calcium magnesium carbonate

The synthesis of ACMC material with 47.9 ± 1.5 mol% Mg was carried out by a previously developed protocol.^{33,41} Briefly, a 0.25 M (Ca, Mg) Cl_2 solution (Mg/Ca ratio = 2.1) and a 0.25 M Na_2CO_3 solution were prepared using $\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$, $\text{MgCl}_2 \cdot 6\text{H}_2\text{O}$, and Na_2CO_3 chemicals (p.a. Roth) and ultrapure water (Millipore Integral 3; 18.2 MΩ cm⁻¹). The two stock solutions were cooled in a fridge for 4 hours ($T = 10 \pm 2$ °C). The precipitation of ACMC was induced by pouring 100 mL of the (Ca, Mg) Cl_2 solution into a beaker containing 100 mL of the Na_2CO_3 solution. The ACMC was separated from solution by a 0.2 μm cellulose filter using a suction filtration unit and was subsequently freeze-dried for 12 hours (Virtis Benchtop 3 L). The synthesis protocol was repeated several times until about 20 g of ACMC were produced. The obtained ACMC material was analysed by X-ray diffraction (Fig. S1A†) and was stored in a desiccator with silica gel.

2.2 Experimental setup for ACMC transformation into HMC

Transformation of ACMC in solution was performed at 10, 20, 40, 60 and 80 °C (see Table 1). Initially, 50 mM NaHCO_3 solutions were prepared in 150 mL gas-tight glass bottles which were placed either in a cooling chamber (10 ± 2 °C), a temperature-controlled room (20 ± 2 °C) or in ovens (40 ± 2 °C, 60 ± 2 °C and 80 ± 2 °C) over a period of 24 to 60 h. 52 mL of the pre-cooled/heated NaHCO_3 solution was added into a 150 mL borosilicate glass reactor containing 6.099 g $\text{MgCl}_2 \cdot 6\text{H}_2\text{O}$ to obtain a Mg concentration of 30 mM. 2 mL of the NaHCO_3 – MgCl_2 solution was collected with a pipette to measure its initial Mg and alkalinity concentration (Table 2). The MgCl_2 – NaHCO_3 solutions exhibit an initial pH value of 8.02 ± 0.15 (Table 2). The reactor was placed in an Easy Max System (Max™ 102; Mettler Toledo) in order to control the stirring of the solution at 200 rpm and its temperature at 10.0, 20.0, 40.0, 60.0 or 80.0 ± 0.1 °C (Fig. 1). Subsequently, 1.5 g of the synthesized ACMC was introduced into the buffered NaHCO_3 – MgCl_2 solution (Fig. 1). After a short homogenization period between the ACMC material and the solution, a homogeneous sub-sample of the suspension (2 mL) was collected with a pipette ($t = 0.3$ min). Subsequently, the pH of the experimental solution was adjusted to 7.58 ± 0.04 (within ≤ 3 min) by computer-controlled titration of a 2 M HCl solution (Schott; TitroLine alpha plus) (Fig. S2†). Note here that the pH of the solution was adjusted to 7.6 to prevent the precipitation of hydrous Ca- or Mg- carbonate phases, such as monohydrocalcite and nesquehonite.^{36,42,43} Further homogeneous sub-samples of the experimental suspension (2 mL) were collected with a pipette at 5 min, 11



Table 1 Mineralogical composition of the solids obtained at certain reaction times during experiments performed from 10 to 80 °C. The quantification of high Mg-calcite (HMC), aragonite (ARG) and hydromagnesite (HMG) at ≥ 1 day of reaction time was realized by Rietveld refinement. $[Mg]_{XRD}$ of HMC: Mg content of HMC calculated from the d_{104} reflection according to Goldsmith *et al.*,⁴⁴ d_{104} : position of the d_{104} reflection of HMC; $FWHM_{d104}$: left full-width at half-maximum of the d_{104} reflection of HMC

Experiment	Time	Composition	HMC	ARG	HMG	$[Mg]_{XRD}$ of HMC	d_{104}	$FWHM_{d104}$
	min per day		wt%	wt%	wt%	mol%		$^{\circ}2\theta$
T_10 °C	0.3 min	ACMC	—	—	—	—	—	—
	5 min	ACMC	—	—	—	—	—	—
	11 min	ACMC	—	—	—	—	—	—
	21 min	ACMC, HMC_I	—	—	—	I: 4.8	I: 3.0204	—
	32 min	HMC_I, ACMC	—	—	—	I: 5.1	I: 3.0194	—
	61 min	HMC_I, HMC_II	—	—	—	I: 6.2, II: 14.5	I: 3.0163, II: 2.9921	—
	1 day	HMC_I, HMC_II, ARG	I: 56, II: 42	2	0	I: 6.6, II: 15.4	I: 3.0153, II: 2.9896	0.375 ^a
	1 week	HMC_I, HMC_II, ARG	I: 57, II: 41	2	0	I: 7.1, II: 16.3	I: 3.0136, II: 2.9869	0.373 ^a
	2 months	HMC_I, HMC_II, ARG	I: 56, II: 41	3	0	I: 6.4, II: 15.8	I: 3.0157, II: 2.9882	0.380 ^a
T_20 °C	0.3 min	ACMC	—	—	—	—	—	—
	5 min	ACMC	—	—	—	—	—	—
	11 min	ACMC, HMC	—	—	—	11.9	2.9996	—
	21 min	ACMC, HMC	—	—	—	12.8	2.9970	—
	32 min	HMC, ACMC	—	—	—	15.2	2.9902	—
	61 min	HMC, ARG	—	—	—	17.0	2.9848	—
	1 day	HMC, ARG	99	1	0	18.0	2.9820	0.459
	1 week	HMC, ARG	99	1	0	18.3	2.9810	0.475
	2 months	HMC, ARG	98	2	0	18.8	2.9795	0.489
T_40 °C	0.3 min	ACMC	—	—	—	—	—	—
	5 min	ACMC	—	—	—	—	—	—
	11 min	ACMC, HMC	—	—	—	23.0	2.9675	—
	21 min	HMC, ACMC	—	—	—	28.4	2.9516	—
	32 min	HMC	—	—	—	29.9	2.9471	—
	61 min	HMC	—	—	—	29.7	2.9477	—
	1 day	HMC	100	0	0	30.3	2.9460	0.644
	1 week	HMC, HMG	94	0	6	29.7	2.9479	0.596
	2 months	HMC, HMG	91	0	9	30.9	2.9443	0.544
T_60 °C	0.3 min	ACMC	—	—	—	—	—	—
	5 min	ACMC	—	—	—	—	—	—
	11 min	ACMC, HMC	—	—	—	34.2	2.9346	—
	21 min	HMC, ACMC	—	—	—	39.4	2.9197	—
	32 min	HMC	—	—	—	39.9	2.9181	—
	61 min	HMC	—	—	—	39.8	2.9183	—
	1 day	HMC	100	0	0	40.7	2.9159	0.659
	1 week	HMC	100	0	0	40.5	2.9163	0.550
	2 months	HMC	100	0	0	40.6	2.9159	0.530
T_80 °C	0.3 min	ACMC	—	—	—	—	—	—
	5 min	ACMC	—	—	—	—	—	—
	11 min	HMC, HMG, ACMC	—	—	—	39.6	2.9189	—
	21 min	HMC, HMG, ARG	—	—	—	39.2	2.9201	—
	32 min	HMC, HMG, ARG	—	—	—	39.5	2.9193	—
	61 min	HMC, HMG, ARG	—	—	—	39.7	2.9187	—
	1 day	HMC, HMG, ARG	78	7	15	39.7	2.9187	0.351
	1 week	HMC, HMG, ARG	80	7	13	40.4	2.9167	0.354
	2 months	HMC, ARG, ^b	—	—	—	41.6	2.9132	0.362

^a Average FWHM of HMC_I and HMC_II. ^b Additional phase that could correspond to magnesite.

min, 21 min, 32 min and 61 min. After 61 min of reaction time, the experimental solutions were transferred into 100 mL gastight bottles which were again placed in the fridge (10 ± 2 °C), temperature-controlled room (20 ± 2 °C) and ovens (40 ± 2 °C, 60 ± 2 °C and 80 ± 2 °C) over a period of 2 months. Additional homogeneous sub-samples were collected after 1 day, 1 week and 2 months of reaction time. The solids were separated from the solutions by a 0.2 μ m cellulose acetate filter using a suction filtration unit and washed with ethanol. Subsequently, the solids were

dried overnight in an oven at 40 °C and stored in a desiccator with silica gel.

2.3 Analytical procedures

Thermogravimetric analyses (TGA) of the freeze-dried ACMC was realized using a PerkinElmer STA8000. The ACMC was heated from 25 °C to 800 °C at 10 °C min⁻¹ in the presence of 99.999% N₂ atmosphere. The mineralogy of the precipitates collected during the ACMC transformation



Table 2 Chemical composition of experimental solutions and solids obtained during experiments performed from 10 to 80 °C. [Ca]_{aq} and [Mg]_{aq}: Ca and Mg concentration of the experimental solution; [Mg]_{solid}: Mg content of the bulk solid determined by chemical digestion (see eqn 1)

Experiment	Time	pH	Alkalinity	[Ca] _{aq}	[Mg] _{aq}	[Mg] _{solid}
	min per day		mM	mM	mM	mol%
T ₁₀ °C	0 min	8.15	49.7	0.0	29.0	47.9 ± 1.5
	0.3 min	9.15	80.6	2.5	41.7	46.0 ± 1.5
	5 min	7.52	126.1	35.1	84.3	38.0 ± 1.4
	11 min	7.55	138.8	41.3	94.3	35.4 ± 1.4
	21 min	7.31	134.3	24.5	110.1	25.5 ± 1.1
	32 min	7.59	137.8	20.1	118.1	22.3 ± 1.0
	61 min	7.53	133.8	12.9	125.2	11.4 ± 0.6
	1 day	7.31	109.1	2.3	122.0	13.0 ± 0.7
	1 week	7.52	97.3	0.6	120.3	14.4 ± 0.7
	2 months	7.77	89.9	0.3	120.7	15.5 ± 0.8
T ₂₀ °C	0 min	8.17	48.4	0.0	29.1	47.9 ± 1.5
	0.3 min	9.21	77.5	2.1	41.4	45.6 ± 1.5
	5 min	7.62	119.4	25.1	81.8	38.0 ± 1.4
	11 min	7.53	126.1	20.8	92.8	34.7 ± 1.4
	21 min	7.59	129.5	16.7	105.6	29.5 ± 1.2
	32 min	7.59	134.2	15.7	115.1	21.9 ± 1.0
	61 min	7.53	118.3	6.9	117.7	16.6 ± 0.8
	1 day	7.49	96.2	0.9	114.2	19.0 ± 0.9
	1 week	7.80	88.2	0.3	111.6	20.8 ± 1.0
	2 months	7.92	79.7	0.2	110.4	22.2 ± 1.0
T ₄₀ °C	0 min	8.01	49.7	0.0	29.0	47.9 ± 1.5
	0.3 min	8.85	70.2	1.7	38.3	47.0 ± 1.5
	5 min	7.59	102.1	14.2	75.3	39.3 ± 1.4
	11 min	7.60	108.3	10.5	85.5	35.5 ± 1.4
	21 min	7.59	110.4	7.8	97.1	30.4 ± 1.3
	32 min	7.55	99.5	3.1	99.0	28.8 ± 1.2
	61 min	7.58	86.4	1.1	97.0	29.6 ± 1.2
	1 day	7.57	60.9	0.4	85.1	33.0 ± 1.3
	1 week	8.52	12.0	0.3	65.2	40.1 ± 1.4
	2 months	8.39	8.1	0.3	64.8	41.2 ± 1.5
T ₆₀ °C	0 min	8.00	49.6	0.0	29.2	47.9 ± 1.5
	0.3 min	8.35	60.4	1.7	32.9	47.5 ± 1.5
	5 min	7.60	80.2	6.6	58.9	43.0 ± 1.5
	11 min	7.59	83.3	5.5	65.6	41.1 ± 1.5
	21 min	7.59	73.9	3.9	67.1	40.5 ± 1.4
	32 min	7.55	57.7	1.1	64.2	40.4 ± 1.4
	61 min	7.57	45.5	0.6	61.3	41.2 ± 1.5
	1 day	7.46	20.1	0.6	48.9	43.8 ± 1.5
	1 week	7.46	7.5	0.5	43.3	45.2 ± 1.5
	2 months	7.64	3.1	0.8	42.2	45.3 ± 1.5
T ₈₀ °C	0 min	7.79	48.5	0.0	29.4	47.9 ± 1.5
	0.3 min	7.82	51.3	1.6	28.1	48.3 ± 1.5
	5 min	7.59	53.7	3.2	33.1	47.6 ± 1.5
	11 min	7.49	39.7	1.4	25.0	48.8 ± 1.5
	21 min	7.41	20.7	0.3	21.8	49.0 ± 1.5
	32 min	7.42	16.7	0.3	20.4	49.3 ± 1.5
	61 min	7.57	10.7	0.3	17.3	49.5 ± 1.5
	1 day	7.87	5.6	0.3	15.7	49.8 ± 1.5
	1 week	7.87	6.3	0.3	16.0	49.6 ± 1.5
	2 months	7.54	6.7	0.9	15.3	49.8 ± 1.5

experiments was determined by a PANalytical X'Pert PRO diffractometer using Co-K α radiation (40 mA, 40 kV) at a 2 θ range from 4 to 85° with a step size of 0.008° 2 θ and 40 s count time per step. The mineral phases of the reaction products were quantified using Rietveld refinement (software HighScore Plus, PANalytical) using the crystal structure data given in Table S1.† The estimated analytical uncertainty of the quantification lies within 1 wt%. The refinement values

consisted of the specimen displacement, a 7-term polynomial background function and scale factors and lattice parameters for all phases. For major phases, the peak shape was refined using a pseudo-Voigt profile function and peak asymmetry. The preferred orientation of calcite was refined for the (104) plane with the March–Dollase function. If two separate calcite phases were present, all their refined values, except of the scale factor and lattice parameters, were constrained with each other to ensure a stable refinement. The d -value and the left full-width at half-maximum (left FWHM, for asymmetric peaks) of the (104) reflection (hereafter referred to as d_{104} and FWHM $_{d104}$, respectively) was obtained from the refinement results. The d_{104} value of the calcite phases was used to calculate their Mg content ([Mg]_{XRD}) according to Goldsmith *et al.*⁴⁴ and this content was then used to adapt the occupancy of Ca and Mg in the structure data used for the refinement. Note that the accuracy of the determination of the d_{104} value and the FWHM $_{d104}$ is limited by the step size of the XRD measurement (0.008° 2 θ).

Selected precipitates were gold-coated and imaged using a scanning electron microscope (SEM, ZEISS DSM 982 Gemini). The total alkalinity of the experimental solutions was measured by a Schott TitroLine alpha plus titrator using a 10 mM HCl solution with a precision of $\pm 2\%$. The aqueous Ca and Mg concentrations of experimental solutions and solids (digested in 6% HNO₃) were determined using inductively coupled plasma optical emission spectrometry (ICP-OES, Perkin Elmer Optima 8300 DV) with an analytical precision of $\pm 3\%$ for Ca and Mg analyses. The Mg content of the bulk solid ([Mg]_{solid} in mol%, Table 2) was calculated according to the equation:

$$[\text{Mg}]_{\text{solid}} = \frac{[\text{Mg}]_{\text{s}}}{[\text{Mg}]_{\text{s}} + [\text{Ca}]_{\text{s}}} \times 100 \quad (1)$$

where [Mg]_s and [Ca]_s are the measured concentrations of Ca and Mg of the solid (mol kg⁻¹) determined by chemical digestion and ICP-OES analysis.

2.4 Geochemical modeling

The aqueous speciation and ion activities of the reactive solutions were calculated using the PHREEQC computer code with the minteq.v4 database. The apparent solubility product (K) for amorphous calcium magnesium carbonate (ACMC, Ca_{1-x}Mg_xCO₃·nH₂O) and high Mg-calcite (HMC, Ca_{1-x}Mg_xCO₃) was calculated using the equation

$$K = (\text{aCa}^{2+})^{1-x} (\text{aMg}^{2+})^x (\text{aCO}_3^{2-}) \quad (2)$$

where aCa²⁺, aMg²⁺ and aCO₃²⁻ are the activities of Ca²⁺, Mg²⁺ and CO₃²⁻ ions in solution, respectively and x is the mole fraction of Mg in the precipitated solid (for HMC: [Mg]_{XRD} in Table 1; for ACMC: [Mg]_{solid} in Table 2). In the latter case the Mg and Ca content of the solid was measured by acid digestion and subsequent ICP-OES analysis, thus representing bulk composition ion-balanced by cation ions. Although water molecules are present in ACMC, the activity



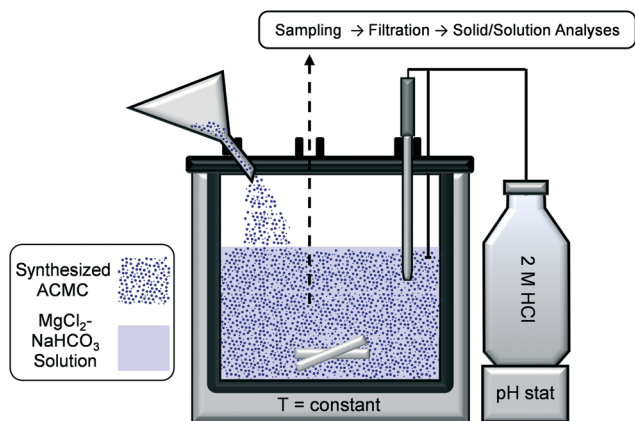


Fig. 1 Experimental setup used for transformation of amorphous calcium magnesium carbonate (ACMC). Synthesized ACMC standard material ($\text{Ca}_{0.52}\text{Mg}_{0.48}\text{CO}_3 \cdot 0.67\text{H}_2\text{O}$) was introduced in a MgCl_2 - NaHCO_3 buffered solution at 10, 20, 40, 60 or 80 °C. Subsequently, the pH of the experimental solution was adjusted to 7.58 ± 0.04 by automatic titration of a 2 M HCl solution.

of water (aH_2O) was not included in eqn 2 (see also Purgstaller *et al.*³³). The activity of water (aH_2O) was assumed to be equal to unity, in agreement with the estimated aH_2O values of 0.995 ± 0.002 of all experimental solutions. It is also worth pointing out that the term solubility product (K) is used herein to be comparable to K values of crystalline phases, although this term is not necessarily valid for amorphous solids.²⁸

3. Results

3.1 Solid phase characterization

Scanning electron microscope observations of the synthesised ACMC (47.9 ± 1.5 mol% Mg) solid reveal spherical particles with <100 nm in size (Fig. S1B†) similar to synthetic amorphous calcium (magnesium) carbonate reported by other authors.^{37,41,45} The water content of the ACMC determined by thermal analysis is 0.67 ± 0.01 moles per unit formula (Fig. S1C†), thus, its stoichiometric formula can be written as $\text{Ca}_{0.52}\text{Mg}_{0.48}\text{CO}_3 \cdot 0.67\text{H}_2\text{O}$. This water content is in good agreement with that of the ACMC ($\text{Ca}_{0.47}\text{Mg}_{0.53}\text{CO}_3 \cdot 0.66\text{H}_2\text{O}$) precipitated by Purgstaller *et al.*³³

X-ray diffraction (XRD) patterns of the solids collected during the ACMC transformation experiments are shown in Fig. 2 at a 2θ range from 33° to 37° , while overview XRD patterns (2θ range = 4 – 65°) are displayed in Fig. S3.† The XRD patterns indicate the sole occurrence of ACMC for 11 min in experiment T₁₀ °C (Fig. 2A and S3A†) and for 5 min in experiments performed at ≥ 20 °C (Fig. 2B–E and S3B–E†). Subsequently, XRD peaks appear due to the transformation of ACMC into high Mg-calcite (HMC) and partly into aragonite (ARG) and hydromagnesite (HMG). XRD analyses reveal that for all experiments at ≤ 60 °C, the reaction products consist mainly of HMC (≥ 98 wt%) and traces of ARG (< 2 wt%) (Table 1). In experiment T₁₀ °C two peaks were detected in the XRD pattern corresponding to high Mg-calcites with distinct Mg contents, hereafter mentioned as HMC_I and HMC_II (Table 1

and Fig. 2A). The reflection peaks of HMC_I appear in the XRD pattern at ≥ 21 min (during ACMC transformation), while those of HMC_II can be detected at ≥ 61 min of reaction time (Fig. 2A and S3A†). Note that in experiment T₄₀ °C, minor amounts of HMG (9 wt%) were detected beside HMC and ARG after 2 weeks of reaction time (Table 1, Fig. S3C†). At 80 °C, XRD patterns show the transformation of ACMC into HMC, HMG and ARG which account for 78, 15 and 7 wt%, respectively, after 1 day of reaction time (Fig. S3E†; Table 1). After 2 months of reaction time, the presence of HMG was no longer detected (Fig. S3E†). Instead, a reflection peak at 2.77 Å indicates the presence of new phase (labeled as * in Fig. S3E†), which could correspond to magnesite (MgCO_3).² Indeed, earlier studies reported the formation of magnesite by the transformation of pre-existing hydrated Mg-carbonate phases such as hydrogmagnesite and dypingite at elevated temperatures (*e.g.* Montes-Hernandez *et al.*⁴⁶).

Fig. 2F shows that the XRD reflection peaks of the final HMCs are broader in comparison to the reference calcite (Merck). This observation is confirmed by comparing the $\text{FWHM}_{d_{104}}$ values, which shows that the $\text{FWHM}_{d_{104}}$ is higher for the HMCs (0.351 – 0.644 ; Table 1) than for the reference calcite (0.094). The obtained $\text{FWHM}_{d_{104}}$ values of the HMCs however are similar to that determined for the reference Coorong dolomite ($\text{FWHM}_{d_{104}} = 0.649$; Fig. S4A†). Broad XRD reflection peaks are referred to non-ideal crystal structure phenomena which can be caused by small (nanocrystalline) particle sizes^{39,40,47} and inhomogeneities in the Ca and Mg composition⁴⁸ of the crystalline solid. Indeed, the broad XRD peaks obtained for the HMCs and the reference Coorong dolomite (Fig. 2F) can result from their small crystal sizes: SEM images reveal aggregates composed of nanocrystals with <100 and <300 nm in size for the HMCs (Fig. 3D) and the Coorong dolomite (Fig. S4B†), respectively.

To get insights into the evolution of the Mg content of the forming HMC *via* ACMC, the characteristic d_{104} -peak value of HMC was used to calculate its Mg content ($[\text{Mg}]_{\text{XRD}}$) according to Goldsmith *et al.*⁴⁴ (Table 1). The Mg content of the HMCs obtained after 1 day of reaction time is higher at elevated temperature as indicated by lower d_{104} -values (corresponds to higher 2θ values in Fig. 2F). Note that a maximum concentration of 40 ± 1 mol% Mg in HMC was obtained in experiments T₆₀ °C and T₈₀ °C (Table 1). In experiment T₁₀ °C, the HMC_I that was detected in the samples collected at ≥ 21 min exhibits 6 ± 1 mol% Mg, which is ~ 9 mol% lower than the Mg content of HMC_II (15 ± 1 mol%) that was detected in the samples obtained at ≥ 61 min of reaction time (Fig. 2A, Table 1). Rietveld refinement of the reaction products after 1 day shows that the amount of HMC_I (7 mol% Mg) and HMC_II (15 mol% Mg) is 56 and 42 wt%, respectively (Table 1). In experiments performed between 20 °C and 60 °C, the initial HMC phase that was detected beside ACMC (at 11 min in Fig. 2B–E; Table 1) shows a significant lower Mg content than the HMC that was obtained during ACMC transformation (Table 1). This indicates that the Mg uptake in HMC is not uniform



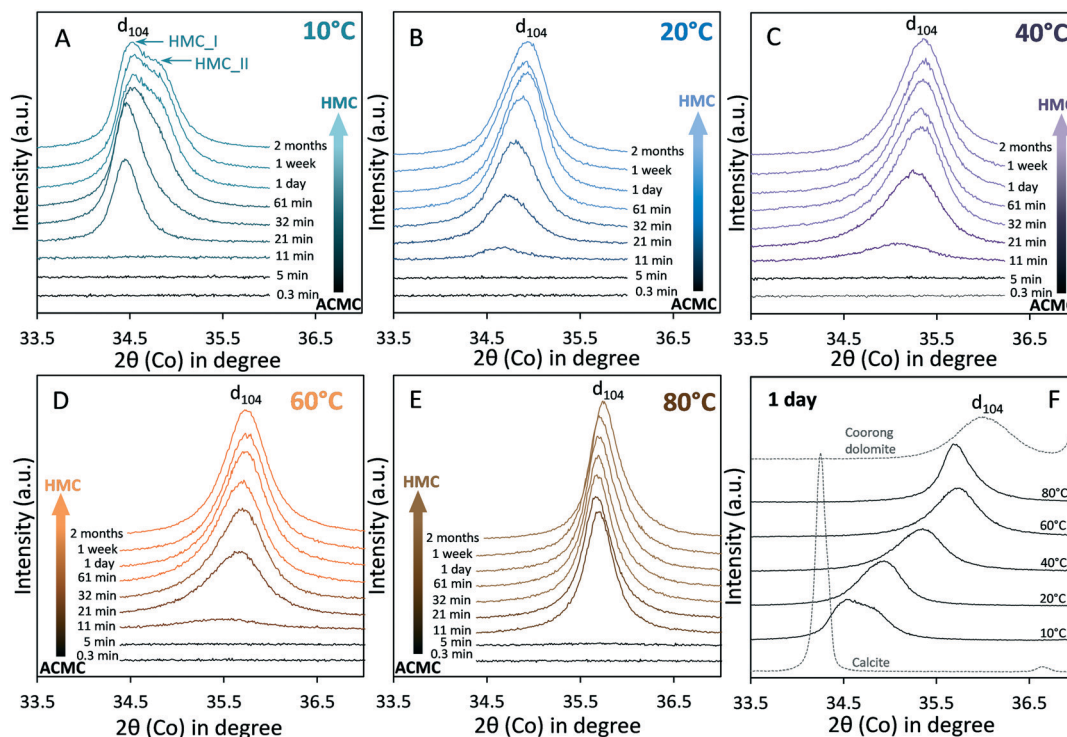


Fig. 2 X-ray diffraction pattern showing the evolution of the d_{104} peak reflection of high Mg-calcite (HMC) formed via transformation of amorphous calcium magnesium carbonate (ACMC) at (A) 10 °C, (B) 20 °C, (C) 40 °C, (D) 60 °C and (E) 80 °C. Xrd pattern from 4° to 65° 2θ are shown in Fig. S3†. Note that the d_{104} -value of the final HMC (F) is generally decreasing at higher temperature (increasing 2θ values) due to elevated Mg content (Goldsmith *et al.*⁴⁴). The X-ray diffraction peaks of the HMCs are broader in comparison to the reference calcite (Merck), but are similar to the reference Coorong dolomite (South Australia).

throughout the transformation process but ends up in HMC with distinct Mg content. In contrast, in experiment T₈₀ °C, the HMCs show no significant change in the $[Mg]_{XRD}$ values during the experimental run (Table 1).

Scanning electron microscope images show that in experiments performed at ≤ 20 °C, the precipitates obtained at 5 min of reaction time consist of amorphous spherical particles (Fig. 3A), similar to the synthesized ACMC standard material (Fig. S1B†). In experiments performed ≥ 40 °C, traces of HMC or HMG nanocrystals were observed beside the ACMC particles at 5 min of reaction time (Fig. 3A), although XRD results show that the solids obtained at 5 min are solely amorphous (Fig. 2C–E). The HMC (T₄₀ °C and T₆₀ °C) consists of spindle-shaped aggregates composed of nanocrystals, whereas the HMG crystals (T₈₀ °C) show a plate-like morphology. These morphologies are similar to those observed for HMC and HMG obtained at a later stage in the experimental runs (Fig. 3B–D). At a reaction time of 21 min (Fig. 3B), large amounts of spherical ACMC particles were observed in the precipitates obtained in experiments performed at ≤ 20 °C, whereas at ≥ 40 °C ACMC was rarely observed in the collected solids. Note here that no ACMC particles were observed in the collected solids at ≥ 21 min in experiment T₈₀ °C (Fig. 3B), at ≥ 32 min in experiments T₄₀ °C and T₆₀ °C (not shown) and at ≥ 61 min in experiments T₁₀ °C and T₂₀ °C (Fig. 3C).

SEM images of the final HMC show aggregates composed of spindle-shaped assemblages of nanocrystals at lower temperatures (≤ 20 °C), whereas at higher temperatures (≥ 40 °C) the HMC-nanocrystals coalesced into spheroidal aggregates (Fig. 3C and D). These spindle-shaped and spherical morphologies are consistent with those of high Mg-calcites formed via amorphous precursors from previous studies.^{39,48–51} In experiment T₁₀ °C, large amounts of star-shaped aggregates composed of linearly arranged nanoparticles were detected beside the spindle-shaped HMC aggregates (Fig. 3D). These star-shaped aggregates most likely correspond to HMC_{II} that was detected beside HMC_I in the XRD-pattern at ≥ 61 min in experiment T₁₀ °C (Fig. 2A). Trace amounts of those aggregates could also be observed in the final solids of experiments T₂₀ °C and T₄₀ °C (Fig. 3D). Note here that the aragonite detected in experiments T₁₀ °C, T₂₀ °C and T₄₀ °C (2–7 wt%, Table 1) could not be found in SEM images.

3.2 Chemical evolution of the reactive solutions and solids

The addition of ACMC into the $MgCl_2$ - $NaHCO_3$ solutions induced an increase of pH as well as of Ca, Mg and alkalinity concentrations (at 0.3 min in Fig. 4A–C, Table 2) which is attributed to the dissolution of ACMC. At lower temperatures, higher amounts of ACMC likely dissolved as it is indicated by higher Ca, Mg and alkalinity concentrations at 0.3 min of



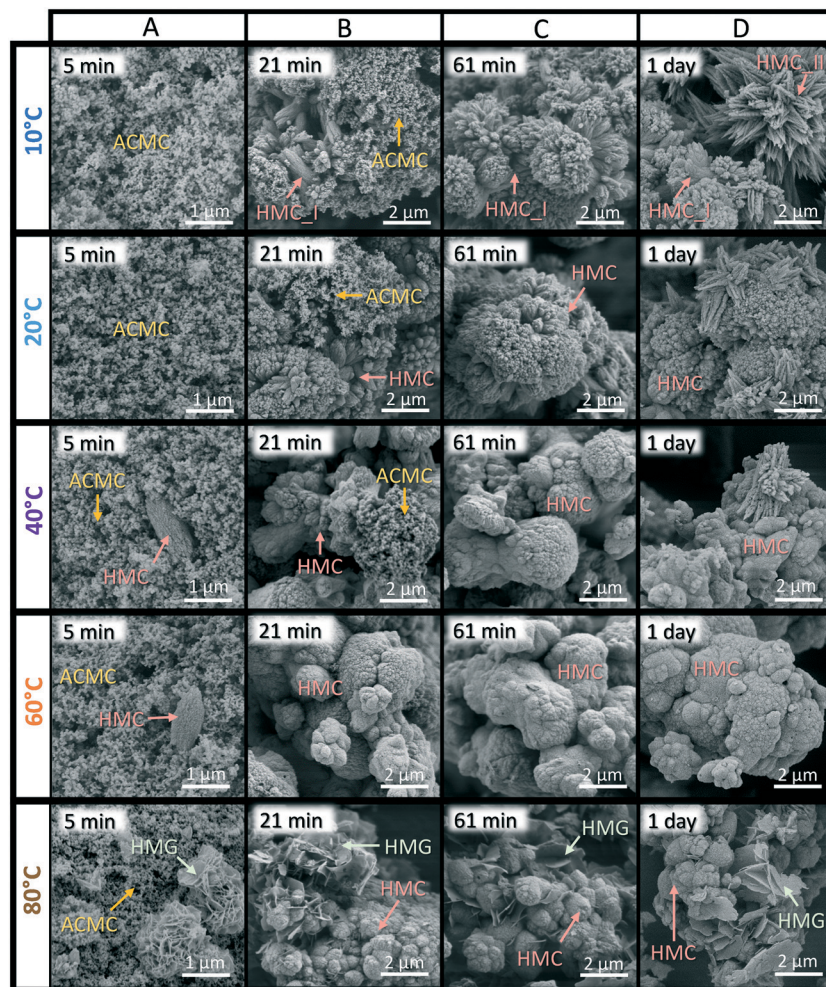


Fig. 3 SEM images of solids collected at 5 min (A), 21 min (B), 61 min (C) and 1 day (D) of reaction time in experiments performed from 10 to 80 °C, showing the transformation of amorphous calcium magnesium carbonate (ACMC) into high Mg-calcite (HMC) at ≤ 60 °C and into HMC and hydromagnesite (HMG) at 80 °C (see Table 1).

reaction time compared to higher temperature experiments (Fig. 4A–C, Table 2). Moreover, the obtained data from experiments performed at lower temperatures show a preferential release of Mg ions into the solution, resulting in ACMCs with slightly lower Mg contents (*e.g.* 46 mol% Mg in T₁₀ °C) compared to original composition of the synthetic ACMC material introduced in the reactors (48 mol% Mg) (Fig. 4D).

After a short initial homogenization period between the ACMC material and the solution, the pH of the experimental solutions was adjusted to 7.58 ± 0.04 by automatic titration of HCl solution (Fig. S2†). The adjustment of the pH resulted in the further dissolution of ACMC, which causes an increase in alkalinity, Ca and Mg concentrations of the experimental solutions between 0.3 and 5 min (Fig. 4A–C). After about 5 min, the Ca and alkalinity concentrations of the experimental solutions decrease (Fig. 4A and C) due to the transformation of ACMC into crystalline Ca–Mg carbonate phases (Fig. 2). With the exception of experiment T₈₀ °C, the aqueous Mg

concentrations of the solutions increase in the same time frame (Fig. 4B). The net release of Mg into the solution with an uptake of Ca into the solid during the ACMC transformation step results in HMCs with lower Mg contents compared to that of the initial ACMC precursor in experiments performed at ≤ 60 °C (Fig. 4D). In contrast, in experiment T₈₀ °C, the Mg content of the bulk solid increases slightly during the ACMC transformation step (Fig. 4D), due to the formation of HMG ($\text{Mg}_5(\text{CO}_3)_4(\text{OH})_2 \cdot 4\text{H}_2\text{O}$) beside HMC and ARG (Table 1, Fig. S3†). Note that in experiment T₄₀ °C, the Mg content of the bulk solid increased between 1 week and 2 months of reaction time, owing to the additional formation of HMG at this stage of the experiment (Fig. 4D).

At higher temperatures, the transformation of ACMC into HMC occurs faster as revealed by XRD and SEM observations (Fig. 2 and 3). This is further supported by the temporal evolution of pH and HCl solution addition derived from titration data (Fig. S2†). In experiments performed at ≤ 60 °C,



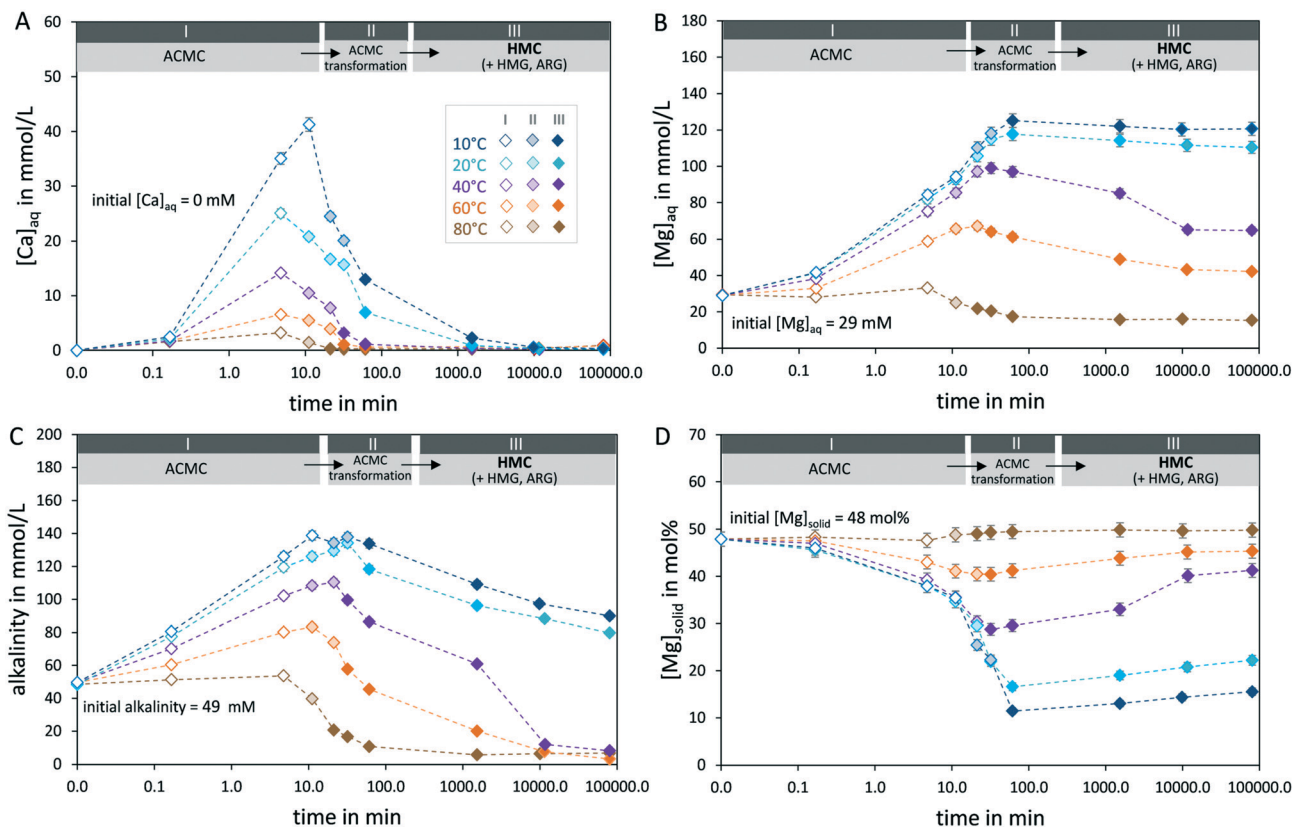


Fig. 4 Concentration of calcium (A), magnesium (B), alkalinity (C) of the solution and Mg content of the bulk solid (D) (see eqn 1) during amorphous calcium magnesium carbonate (ACMC) transformation into high Mg-calcite (HMC) and in case partly into hydromagnesite (HMG) and aragonite (ARG) (see Table 1) at 10, 20, 40, 60 and 80 °C. Stage I: ACMC is the sole solid phase (open diamonds); stage II: ACMC transforms to the crystalline phases (shaded diamonds); stage III: solid phase is crystalline (solid diamonds). Note that the pH of the solution was adjusted to 7.58 ± 0.04 within ≤ 3 min by computer-controlled titration of a 2 M HCl solution (Fig. S2†).

our data reveal that additional amounts of HCl solution were automatically added into the experimental solution to keep the pH constant at 7.58 ± 0.04 during the transformation of ACMC into HMC (Fig. S2A–D†). The rise in pH and consequent titration of HCl may be explained by a fast dissolution of ACMC and slow precipitation of HMC during the ACMC transformation step. At higher temperatures, the HCl solution was added into the experimental solution in shorter periods of time (e.g. 17 min in T₆₀ °C) than at lower temperatures (e.g. 34 min in T₂₀ °C), indicating that the transformation of ACMC into HMC occurs faster at elevated temperatures. Note that in experiment T₈₀ °C, where ACMC transformed into HMC, HMG and ARG (Fig. 2E, Table 1), no additional HCl was added into the solution (Fig. S2E†). In the latter case, the pH decreased slightly from 7.59 to 7.41 during the ACMC transformation step.

4. Discussion

4.1 Effect of temperature on the Mg content of HMC formed via ACMC with near-dolomite stoichiometry

Earlier studies showed a strong correlation between the physicochemical parameters of the initial solution used for

ACMC synthesis and the amount of Mg incorporated into ACMC and the final HMC.^{31,37,49,51} In this context, it has been suggested that tuning the amount of Mg in ACMC controls the amount of Mg incorporated into HMC. In contrast, the results of this study show that the Mg content of the HMC is by far not pre-determined by the Mg content of the initial ACMC (48 mol% Mg) that was introduced into the solution (Fig. 5A). In experiments performed at temperatures ≤ 40 °C, the Mg content of the final HMC is significantly lower (e.g. 18 mol% Mg in experiment T₂₀ °C) than of the initial ACMC, whereas at higher temperatures (≥ 60 °C), the final HMCs reach Mg contents comparable to those of VHMC (40 mol%; Fig. 5A). These findings clearly document that the Mg content of the forming calcite is strongly regulated by the reaction temperature prevailing during ACMC transformation. It is generally acknowledged that the transformation mechanism of ACMC into (V)HMC involves (i) dissolution of ACMC and (ii) nucleation and crystal growth of (V)HMC.^{31,36,40,51} The relationship in Fig. 5A suggests that the Mg content of calcite forming at the ACMC-solution interface is significantly controlled by the temperature-dependent dehydration kinetic of aqueous Mg^{2+} . The incorporation of Mg into anhydrous carbonates is generally postulated to be kinetically limited under low-



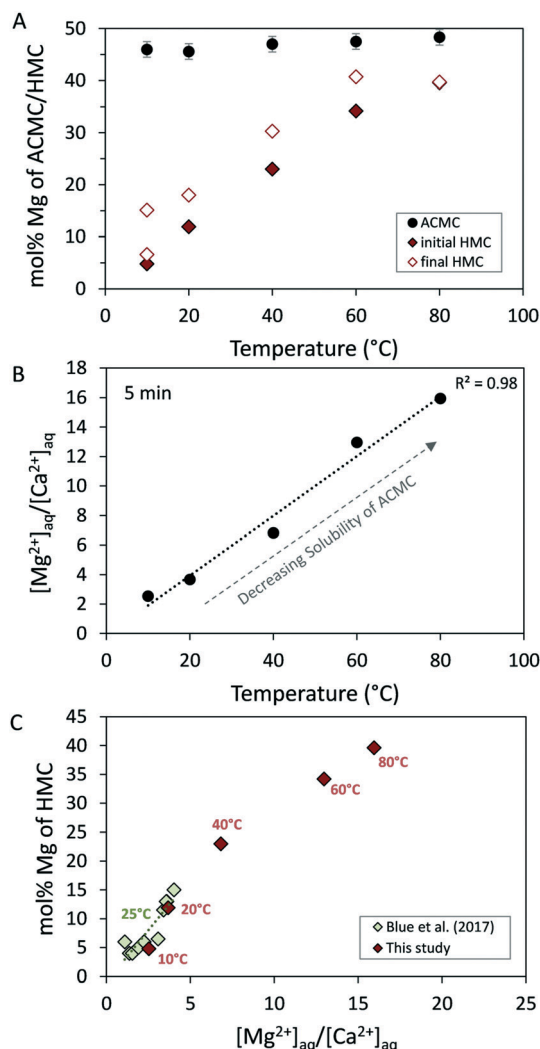


Fig. 5 (A) Temperature of the solution versus the Mg content of (i) ACMC obtained at 0.3 min of reaction time ($[\text{Mg}]_{\text{solid}}$ values in Table 2), (ii) initial HMC obtained at 21 min in experiment T_{10} °C and 11 min in experiments performed at ≥ 20 °C and (iii) final HMC obtained at 1 day of reaction time. Note here that the Mg contents of HMCs were calculated from the d_{104} values of calcite XRD peaks ($[\text{Mg}]_{\text{XRD}}$ values in Table 1), whereas the Mg content of ACMC was calculated from solid digestion ($[\text{Mg}]_{\text{solid}}$ values in Table 2). (B) Temperature of the solution versus the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution at 5 min of reaction time (Table S2†). (C) $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution at 5 min versus the Mg content of (i) initial HMC from this study, of initial HMC from experiments conducted at stirred conditions reported by Blue *et al.*⁵¹

temperature conditions, due to the strong hydration of Mg^{2+} compared to Ca^{2+} .^{20,52} In contrast, higher temperatures reduce the Mg-solvation energy barrier and favour anhydrous Ca–Mg-carbonate formation.⁵³ Thus, the amount of Mg incorporated into calcite formed *via* ACMC is enhanced at higher reaction temperatures (Fig. 5A).

The obtained results further support previous observations⁵¹ that the incorporation of Mg into calcite formed *via* ACMC is controlled by the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the reactive solution at the time of transformation. Blue *et al.*⁵¹ synthesized ACMCs with different Mg contents under

constant pH conditions (pH 8.3–9.1) using a mixed flow reactor and investigated their crystallization pathway in aqueous solution under stirred and unstirred conditions. Their results document a positive correlation between the Mg content of calcite and the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution from which the mineral formed. In the experimental setup used in the present study, Mg^{2+} and CO_3^{2-} ions were present in the initial solution, whereas Ca^{2+} ions exclusively originated from the dissolution of the ACMC. At higher temperatures, lower amounts of ACMC dissolved in the NaHCO_3 – MgCl_2 solution as indicated by less Ca^{2+} ions released into the reactive solution (Fig. 4A). This can be explained by the anticorrelation between amorphous carbonate mineral solubility and temperature (for ACC: Brečević & Nielsen²⁸). Thus, at 5 min of reaction time, the reactive solutions exhibit higher aqueous $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratios as a function of increasing temperature (Fig. 5B; Table S2†). Indeed, by comparing the Mg content of the initial HMC with the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution after its reaction with the ACMC, we find a good agreement of the results obtained at 25 °C and those reported by Blue *et al.*⁵¹ (Fig. 5C).

During the ongoing transformation of ACMC into HMC, the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the solution increases (Table S2†) due to stronger uptake of Ca^{2+} compared to Mg^{2+} from the solution into the solid. This increasing $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio likely results in the formation of HMC with higher Mg contents as a function of reaction time ($[\text{Mg}]_{\text{XRD}}$ values in Table 1). Hence, the final HMCs exhibit higher Mg contents compared to the initial HMCs (Fig. 5A). This effect is temporally better resolved at low temperatures, due to slower transformation kinetics of ACMC into HMC (Table 1). For example, at 20 °C the Mg content of HMC increases slowly from 12 to 17 mol% between 11 and 61 min of reaction time (Table 1). In contrast, at higher temperatures, the transformation of ACMC into HMC occurs faster and the transition from lower to higher Mg contents for HMCs occurs in shorter time intervals (Table 1).

The positive relationship between aqueous $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratios and Mg content of the forming HMC from ACMC transformation (Fig. 5C) indicates that the Mg content into calcite might be enhanced to higher levels by increasing the $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the reactive transformation solution. However, experimental studies at 25 °C and moderate pH conditions (8.3–9.1) documented that at higher prevailing $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratios (>4 : Blue *et al.*⁵¹; ≥ 8 : Purgstaller *et al.*⁴²), ACMC transforms into hydrous Ca- or Mg-carbonate phases, *i.e.* monohydrocalcite and nesquehonite, rather than into calcite, due to the inhibition of the latter.^{50,51} These observations suggest a maximum limit of about 20 mol% Mg in calcite formed *via* the amorphous pathway.^{49–51} It should be emphasized that when ACMC transforms under unstirred conditions, Mg contents of up to 30 mol% Mg for HMC were reported in experimental studies conducted at moderate pH conditions and ambient temperatures.^{51,54} However the process by which physical mixing influences the formation of distinct crystalline Ca–Mg-carbonates and the Mg content of calcite is still poorly understood and needs further investigations.



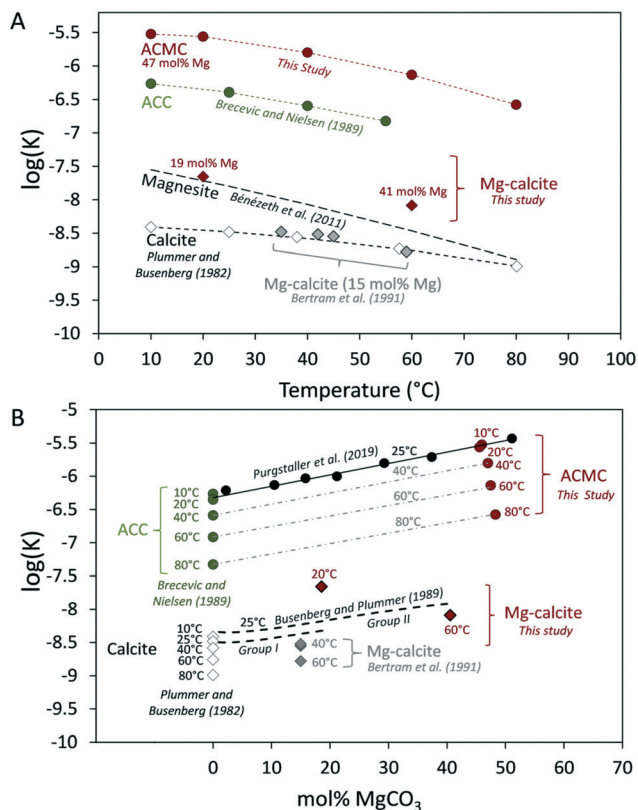


Fig. 6 Solubility product (K) of ACC (Brečević and Nielsen²⁸), ACMC (present study and Purgstaller *et al.*³³), calcite (Plummer and Busenberg⁵⁵), Mg-calcite (present study, Busenberg and Plummer¹⁹ and Bertram *et al.*⁵⁷) and magnesite (Bénézeeth *et al.*⁵⁶) as a function of (A) temperature and (B) Mg mole fraction. The K of ACMC and Mg-calcite was calculated according to eqn 2.

4.2 Solubility considerations of ACMC and HMC

At higher reaction temperatures, less ACMC dissolved in the NaHCO₃-MgCl₂ solutions as it is indicated by the lower concentrations of Ca, Mg and CO₃ ions in the reactive solutions at 0.3 min of reaction time (Fig. 4A). In order to assess the effect of temperature on the solubility (K) of ACMC, the activities of Ca²⁺, Mg²⁺ and CO₃²⁻ ions in the reactive solution (Table S3†) and the ACMC stoichiometry (at 0.3 min) were used to calculate K_{ACMC} values according to eqn 2 (see also Purgstaller *et al.*³³). The obtained K_{ACMC} values for ACMC with 47.0 ± 1.5 mol% Mg ($t = 0.3$ min) indicate that the solubility of ACMC is lower at higher temperature (Fig. 6A), analogous to the temperature-dependent solubility of ACC determined by Brečević and Nielsen.²⁸ A similar behaviour is also well known for anhydrous carbonate minerals such as calcite⁵⁵ and magnesite⁵⁶ which exhibit lower K values at higher temperatures (Fig. 6A). The relationship between the solubility of ACMC (with 47.0 ± 1.5 mol% Mg) and temperature (T in Kelvin) can be described by the polynomial function:

$$\log(K_{\text{ACMC}}) = -0.0001434 \pm 0.0000137 T^2 + 0.0760 \pm 0.0087 T - 15.537 \pm 1.375 \quad (R^2 = 0.99) \quad (3)$$

It is strongly emphasized that in the present study the experimental data for calculating K_{ACMC} values are limited (Table S3†) and further experimental work is needed to verify the obtained relationship between K_{ACMC} and temperature (Fig. 6A).

As it can be seen in Fig. 6A the solubility products of ACMC are higher compared to Mg-free ACC at any temperature suggesting that ACMC is more soluble than ACC. This feature can be attributed to the increasing water content and changes in short-range order, as Ca is substituted by Mg in the amorphous structure.³³ The relationship between the solubility of ACMC and its Mg content is displayed in Fig. 6B. Note that the slopes of the regression lines between the K_{ACC} value of Brečević and Nielsen²⁸ and the K_{ACMC} value of this study are similar at 10, 20, 40, 60 and 80 °C (slope = 0.01634 ± 0.00066; Fig. 6B and Table S4†) and are in good agreement with the slope value of the linear relationship between K_{ACMC} and the Mg content of ACMC at 25 °C (slope = 0.01602; Mg content of ACMC = 2 to 51 mol%, Table S4†) presented by Purgstaller *et al.*³³ Considering the close agreement between K values obtained in this study and those of Brečević and Nielsen²⁸ and Purgstaller *et al.*³³ a quantitative description of the variations of K_{ACMC} with temperature and Mg content of ACMC can be obtained starting from the relationship between ACC solubility (K_{ACC}) and temperature (T in Kelvin) determined by Brečević and Nielsen:²⁸

$$\log(K_{\text{ACC}}) = -0.0001096 T^2 + 0.0545 T - 12.919 \quad (4)$$

Based on the assumption that the relative increase of K_{ACMC} values as a function of the Mg content is independent of temperature and can be described by the same linear function with an average slope of 0.01629 ± 0.00060 (see Table S4†), the variation of the solubility of ACMC with temperature and Mg content can be obtained from the equation:

$$\log(K_{\text{ACMC}}) = 0.01629 \times [\text{Mg}]_{\text{ACMC}} - 0.0001096 T^2 + 0.0545 T - 12.919 \quad (5)$$

where $[\text{Mg}]_{\text{ACMC}}$ refers to the Mg content of ACMC in mol% (eqn 1) and T is the temperature reported in Kelvin. Note that the obtained relationship between K_{ACMC} , $[\text{Mg}]_{\text{ACMC}}$ and temperature is only valid for the studied ranges of $[\text{Mg}]_{\text{ACMC}} = 0\text{--}51$ mol% and $T = 10\text{--}80$ °C of the present study, Brečević and Nielsen²⁸ and Purgstaller *et al.*³³

The dependence of ACMC solubility on the Mg content is similar to that of Mg-calcite (above 4 mol% Mg), where high Mg contents in calcite result in higher solubilities (for 25 °C: Busenberg and Plummer;¹⁹ Fig. 6B). However, the impact of temperature on the solubility of HMC is not fully elucidated and to our knowledge only Bertram *et al.* (1991)⁵⁷ studied the solubility of synthetic Mg-calcites with 1.9 to 15 mol% Mg between 25 to 64 °C and showed that the decrease in solubility of Mg-calcite with increasing temperature parallels



that of calcite (see K data of Mg-calcite with 15 mol% Mg in Fig. 6A). No experimental data, however, exist for the solubility dependence of HMC with >15 mol% Mg at elevated temperatures. Fig. 6A and B shows the calculated K values of the final HMC (K_{HMC}) from experiment T₂₀ °C and T₆₀ °C. The K -values were calculated using the chemical data and the HMC stoichiometry (based on $[\text{Mg}]_{\text{XRD}}$, Table 1) at 1 week and 2 months of reaction time (Table S3†). At this time the system achieved near-equilibrium conditions (Table 2). Note that K_{HMC} values were not calculated in experiments T₁₀ °C, T₄₀ °C and T₈₀ °C because the final solids of T₁₀ °C consist of two Mg-calcites with different Mg contents (Fig. 2A) and those of T₄₀ °C and T₈₀ °C include HMG (6–15 wt%, Table 1). As it can be seen in Fig. 6B, the HMC formed *via* ACMC at 20 °C (with 18 mol% Mg) is significantly more soluble ($\log(K_{\text{HMC}}) = -7.83$) than the biogenic and synthetic Mg-calcites of group II reported by Busenberg and Plummer,¹⁹ which exhibit similar Mg contents (*e.g.* $\log(K_{\text{HMC}}) = -8.16$ for biogenic Mg-calcite with 19 mol% Mg at 25 °C). Note that Busenberg and Plummer¹⁹ distinguished two groups of Mg-calcites with distinct solubilities, reflecting crystal structure defects (Fig. 6B). Group I consists of metamorphic and hydrothermal Mg-calcites and synthetic Mg-calcites prepared at high temperatures and pressures or low temperatures and low supersaturations (saturation index ≤ 0.2). Group II Mg-calcites are of either biogenic origin or synthetic Mg-calcites prepared at high supersaturations (saturation index ≥ 0.5). The Mg-calcites of group I are thought to be relatively free of structural defects, whereas group II Mg-calcites were described as more defective Mg-calcites which exhibit higher solubilities. The discrepancy between the reported solubility value of Busenberg and Plummer¹⁹ and of the present study might be explained by different physical defects originating from rapid crystal growth during HMC formation. Considering an ACMC transformation step, the solution in equilibrium with ACMC is highly supersaturated with respect to calcite (saturation index >2 (ref. 31) and 49)). Such high calcite supersaturation degrees probably promote rapid precipitation of HMC at the ACMC-solution interface. Indeed, in the present study, ACMC transformed into nanocrystalline HMC at relatively short time intervals (<60 min, Fig. 2 and 3). These solids are likely more defective and thus exhibit higher solubilities compared to well-crystallized HMC (Fig. 6B). The elevated solubilities of nanocrystalline HMC appears to be consistent with recent results of Davis *et al.*⁵⁸ who demonstrated, based on microscopic observations, enhanced solubilities of Mg-calcites at the nanometric scale.

5. Conclusions

In the present study, the transformation of ACMC with near-dolomite stoichiometry (47.9 ± 1.5 mol% Mg) into HMC was investigated in a MgCl_2 – NaHCO_3 buffered solution at pH 7.6 and in the temperature range from 10 to 80 ± 1 °C. The results provide an advanced understanding of the complex

interplay between the chemical composition of ACMC, the corresponding solution and the finally formed Mg-calcite. At first, the experimental data indicate that the apparent solubility of ACMC is lower at higher temperature and the relative increase of solubility as a function of the Mg content of ACMC is similar from 10 to 80 °C which can be assessed by eqn 5. At ambient temperatures, the Mg content of HMC formed *via* the transformation of ACMC is as low as 18 mol%, whereas at 60 and 80 °C, the final HMCs reach Mg levels comparable with those of VHMC (40 mol%). Hence, the Mg content of the HMC is not pre-determined by the Mg content of the amorphous precursor, but strongly influenced by the reaction temperature during ACMC transformation. The transformation of ACMC to HMC proceeds *via* a dissolution and re-precipitation mechanism where at higher temperatures the amount of Mg incorporated into calcite is enhanced due to (i) the reduced solvation energy barrier of aqueous Mg^{2+} and (ii) the high $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the reactive solution after its reaction with the ACMC. The high $\text{Mg}^{2+}/\text{Ca}^{2+}$ ratio of the latter solution can be explained by low solubility of ACMC at elevated temperatures. The obtained solubilities of the final HMC are higher compared to those of well-crystalline Mg-bearing calcites, which is most likely caused by nanometric-sized HMC crystals with structural defects originated from fast crystal growth.

Abbreviations

ACC	Amorphous calcium carbonate
ACMC	Amorphous calcium magnesium carbonate
ARG	Aragonite
FWHM	Full-width at half-maximum
ICP-OES	Inductively coupled plasma optical emission spectrometry
LMC	Low Mg-calcite
HMC	High Mg-calcite
HMG	Hydromagnesite
SEM	Scanning electron microscope
TGA	Thermogravimetric analyses
XRD	X-ray diffraction

Conflicts of interest

There are no conflicts to declare.

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References

- J. W. Morse and F. T. Mackenzie, in *Developments in Sedimentology* 48, Elsevier, 1990, p. 725.
- R. J. Reeder, in *Reviews in Mineralogy*, Mineralogical Society of America, 11th edn, 1990, p. 399.
- F. T. Mackenzie, W. D. Bischoff, F. C. Bishop, M. Loijens, J. Schoonmaker and R. Wollast, *Carbonates Mineral. Chem.*, 1983, pp. 97–144.
- W. D. Bischoff, S. K. Sharma and F. T. MacKenzie, *Am. Mineral.*, 1985, **70**, 581–589.
- J. M. Gregg, D. L. Bish, S. E. Kaczmarek and H. G. Machel, *Sedimentology*, 2015, 1749–1769.
- M. E. Tucker and V. P. Wright, *Carbonate Sedimentology*, Blackwell Scientific Publications, Boston, 1990.
- V. Mavromatis, R. Botz, M. Schmidt, V. Liebetrau and C. Hensen, *Int. J. Earth Sci.*, 2014, **103**, 1831–1844.
- A. Paul and S. W. Lokier, *Sediment. Geol.*, 2017, **352**, 1–13.
- S. Bentov and J. Erez, *Geochem., Geophys., Geosyst.*, 2006, **7**, Q01P08.
- S. Raz, S. Weiner and L. Addadi, *Adv. Mater.*, 2000, **12**, 38–42.
- M. C. Nash, U. Troitzsch, B. N. Opdyke, J. M. Trafford, B. D. Russell and D. I. Kline, *Biogeosciences*, 2011, **8**, 3331–3340.
- S. Gayathri, R. Lakshminarayanan, J. C. Weaver, D. E. Morse, R. Manjunatha Kini and S. Valiyaveetil, *Chem. – Eur. J.*, 2007, **13**, 3262–3268.
- J. H. Schroeder, E. J. Dwornik and J. J. Papike, *Geol. Soc. Am. Bull.*, 1969, **80**, 1613–1616.
- Y. Ma, B. Aichmayer, O. Paris, P. Fratzl, A. Meibom, R. A. Metzler, Y. Politi, L. Addadi, P. U. P. A. Gilbert and S. Weiner, *Proc. Natl. Acad. Sci. U. S. A.*, 2009, **106**, 6048–6053.
- F. C. Furman, R. E. Woody, M. A. Rasberry, D. J. Keller and J. M. Gregg, *Proc. Sixth Symp. Geol. Bahamas*, 1993, 47–54.
- C. Vasconcelos and J. A. McKenzie, *J. Sediment. Res.*, 1997, **67**, 378–390.
- J. B. Sumrall, E. B. Larson and J. E. Mylroie, *Carbonates Evaporites*, 2017, **32**, 123–133.
- D. Fussmann, A. J. E. Von Hoyningen-Huene, A. Reimer, D. Schneider, H. Babková, R. Peticzka, A. Maier, G. Arp, R. Daniel and P. Meister, *Biogeosciences*, 2020, **17**, 2085–2106.
- E. Busenberg and L. Niel Plummer, *Geochim. Cosmochim. Acta*, 1989, **53**, 1189–1208.
- F. Lippmann, *Sedimentary carbonate minerals*, Springer-Verlag Berlin, 1973, vol. 12.
- R. A. Berner, *Geochim. Cosmochim. Acta*, 1975, **39**, 489–504.
- L. Fernandez-Diaz, A. Putnis, M. Prieto and C. V. Putnis, *J. Sediment. Res.*, 1996, **66**, 482–491.
- K. E. Goetschl, B. Purgstaller, M. Dietzel and V. Mavromatis, *Geochim. Cosmochim. Acta*, 2019, **265**, 505–519.
- G. D. Markham, J. P. Glusker, C. L. Bock, M. Trachtman and C. W. Bock, *J. Phys. Chem.*, 1996, **100**, 3488–3497.
- A. Bleuzen, P. A. Pittet, L. Helm and A. E. Merbach, *Magn. Reson. Chem.*, 1997, **35**, 765–773.
- J. Schott, O. Pokrovsky and E. Oelkers, *Rev. Mineral. Geochem.*, 2009, **70**, 207–258.
- T. Ogino, T. Suzuki and K. Sawada, *J. Cryst. Growth*, 1990, **100**, 159–167.
- L. Brečević and A. E. Nielsen, *J. Cryst. Growth*, 1989, **98**, 504–510.
- L. Addadi, S. Raz and S. Weiner, *Adv. Mater.*, 2003, **15**, 959–970.
- Y. Politi, D. R. Batchelor, P. Zaslansky, B. F. Chmelka, J. C. Weaver, I. Sagi, S. Weiner and L. Addadi, *Chem. Mater.*, 2010, **22**, 161–166.
- J. D. Rodriguez-Blanco, S. Shaw and L. G. Benning, *Am. Mineral.*, 2015, **100**, 1172–1181.
- A. V. Radha, A. Fernandez-Martinez, Y. Hu, Y. S. Jun, G. A. Waychunas and A. Navrotsky, *Geochim. Cosmochim. Acta*, 2012, **90**, 83–95.
- B. Purgstaller, K. E. Goetschl, V. Mavromatis and M. Dietzel, *CrystEngComm*, 2019, **21**, 155–164.
- J. D. Rodriguez-Blanco, S. Shaw, P. Bots, T. Roncal-Herrero and L. G. Benning, *J. Alloys Compd.*, 2012, **536**, S477–S479.
- C. J. Lin, S. Y. Yang, S. J. Huang and J. C. C. Chan, *J. Phys. Chem. C*, 2015, **119**, 7225–7233.
- J. M. Xto, H. Du, C. N. Borca, E. Amstad, J. A. Van Bokhoven and T. Huthwelker, *Cryst. Growth Des.*, 2019, **19**, 4385–4394.
- X. Long, Y. Ma and L. Qi, *Cryst. Growth Des.*, 2011, **11**, 2866–2873.
- H. Yang, S. Chai, Y. Zhang and Y. Ma, *CrystEngComm*, 2015, **18**, 157–163.
- M. Schmidt, S. Xeflide, R. Botz and S. Mann, *Geochim. Cosmochim. Acta*, 2005, **69**, 4665–4674.
- I. J. Kelleher and S. A. T. Redfern, *Mol. Simul.*, 2002, **28**, 557–572.
- F. Konrad, F. Gallien, D. E. Gerard and M. Dietzel, *Cryst. Growth Des.*, 2016, **16**, 6310–6317.
- B. Purgstaller, M. Dietzel, A. Baldermann and V. Mavromatis, *Geochim. Cosmochim. Acta*, 2017, **217**, 128–143.
- F. Konrad, B. Purgstaller, F. Gallien, V. Mavromatis, P. Gane and M. Dietzel, *J. Cryst. Growth*, 2018, **498**, 381–390.
- J. R. Goldsmith, D. L. Graf and H. C. Heard, *Am. Mineral.*, 1961, **46**, 453–459.
- C. R. Blue and P. M. Dove, *Geochim. Cosmochim. Acta*, 2015, **148**, 23–33.
- G. Montes-Hernandez, F. Renard, R. Chiriac, N. Findling and F. Toche, *Cryst. Growth Des.*, 2012, **12**, 5233–5240.
- W. Bischoff, F. Bishop and F. Mackenzie, *Am. Mineral.*, 1983, **68**, 1183–1188.
- F. Zhang, H. Xu, H. Konishi and E. E. Roden, *Am. Mineral.*, 2010, **95**, 1650–1656.
- B. Purgstaller, V. Mavromatis, A. Immenhauser and M. Dietzel, *Geochim. Cosmochim. Acta*, 2016, **174**, 180–195.
- B. Purgstaller, F. Konrad, M. Dietzel, A. Immenhauser and V. Mavromatis, *Cryst. Growth Des.*, 2017, **17**, 1069–1078.
- C. R. Blue, A. Giuffre, S. Mergelsberg, N. Han, J. J. De Yoreo and P. M. Dove, *Geochim. Cosmochim. Acta*, 2017, **196**, 179–196.
- V. Mavromatis, Q. Gautier, O. Bosc and J. Schott, *Geochim. Cosmochim. Acta*, 2013, **114**, 188–203.



- 53 F. Di Lorenzo, R. M. Rodríguez-Galán and M. Prieto, *Mineral. Mag.*, 2014, **78**, 1363–1372.
- 54 N. Han, C. R. Blue, J. J. De Yoreo and P. M. Dove, *Procedia Earth Planet. Sci.*, 2013, **7**, 223–227.
- 55 N. Plummer and E. Busenberg, *Geochim. Cosmochim. Acta*, 1982, **46**, 1011–1040.
- 56 P. Bénézech, G. D. Saldi, J. L. Dandurand and J. Schott, *Chem. Geol.*, 2011, **286**, 21–31.
- 57 M. A. Bertram, F. T. Mackenzie, F. C. Bishop and W. D. Bischoff, *Am. Mineral.*, 1991, **76**, 1889–1896.
- 58 K. J. Davis, P. M. Dove and J. J. De Yoreo, *Science*, 2000, **290**, 1134–1137.

