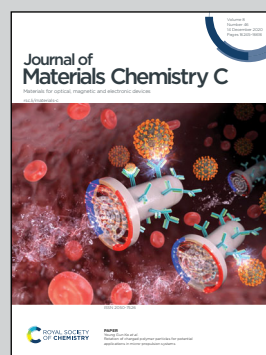


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Universal host materials for red, green and blue high-efficiency single-layer phosphorescent organic light-emitting diodes

High-efficiency molecular designs of universal host materials leading to high-performance red, green and blue single-layer phosphorescent OLEDs are reported. A step towards a universal host for simplified OLEDs. J. F. Bergamini is acknowledged for the image.

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Universal host materials for red, green and blue high-efficiency single-layer phosphorescent organic light-emitting diodes†

Fabien Lucas, ^a Cassandre Quinton,^a Sadiara Fall,^b Thomas Heiser, ^b Denis Tondelier, ^c Bernard Geffroy, ^{cd} Nicolas Leclerc, ^e Joëlle Rault-Berthelot ^a and Cyril Poriel ^{*,a}

Simplifying the structure of Organic Light-Emitting Diodes (OLEDs) has been for the last twenty years the purpose of many studies. However, despite these efforts, only a few materials provide high efficiency devices. We report herein efficient design strategies to construct universal host materials for red, green and blue Single-Layer Phosphorescent OLEDs (SL-PhOLEDs). The three materials investigated, **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂**, have been synthesized via an efficient approach and are constructed on the association of an electron rich phenylacridine unit connected by a spiro carbon atom to three different electron-deficient diphenylphosphineoxide-substituted fluorenes. Electrochemical, spectroscopic, thermal and transport properties are discussed. The position (C2 and C7 vs. C3 and C6) and the number (1 vs. 2) of diphenylphosphineoxide units on the fluorene backbone have been particularly studied to highlight the best combination in term of device performance. Red, green and blue SL-PhOLEDs (RGB SL-PhOLEDs) have been fabricated and characterized and their performances discussed. Of particular interest, we managed to reach a FIr6-based SL-PhOLED (with **SPA-2-FPOPh₂**) possessing an external quantum efficiency of 9.1% and a low threshold voltage (below 3 V). As far as we know, this is the first example of SL-PhOLED using this blue phosphorescent emitter. On the other hand, with notably a very high external quantum efficiency of 18% with FIrpic as sky blue emitter, **SPA-2,7-F(POPh₂)₂** displays the highest overall performance in the series and the highest overall performance ever reported for RGB SL-PhOLEDs using a universal host. This not only shows that the association of phenylacridine and diphenylphosphineoxide units fulfils the required criteria for an universal host for high efficiency SL-PhOLEDs but also highlights that the arrangement of these fragments drives the device performance.

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Introduction

The development of efficient organic host matrices for the emissive layer (EML) of Phosphorescent Organic Light-Emitting Diodes (PhOLEDs)¹ is at the origin of the fantastic progresses

made by this technology in the last twenty years.^{1–4} The role played by the host matrix is crucial as it should prevent energy back transfers from the guest emitter to the host and favour the confinement of excitons. Today, the rational design of host materials for guest phosphors has allowed to reach very high-efficiency red, green or blue PhOLEDs (with external quantum efficiency EQE > 25%).^{5–18} However, all these are multi-layer devices, which are constituted of a stack of organic layers in order to improve the injection, transport and recombination of charges within the EML. To reduce the cost and the environmental footprint of the OLED technology, simplifying the multi-layer structure is one interesting direction for the future.¹⁹ The so-called Single-Layer PhOLEDs (SL-PhOLEDs), the simplest device only made of the electrodes and the EML, have thus stimulated a strong interest for the last fifteen years. However, reaching high efficiency SL-PhOLEDs of all the colours (red,^{20–25} yellow,^{21,26,27} orange,^{26,27} green,^{21,23,26,28–34} blue^{23,24,33,35–37} and white³⁸) is a

^a Univ Rennes, CNRS, ISCR-UMR CNRS 6226, F-35000 Rennes, France.E-mail: cyril.poriel@univ-rennes1.fr^b Laboratoire ICube, Université de Strasbourg, UMR CNRS 7357, 67087 Strasbourg, France^c LPICM, CNRS, Ecole Polytechnique, Institut Polytechnique de Paris, route de Saclay, 91128 Palaiseau, France^d Université Paris-Saclay, CEA, CNRS, NIMBE, LICSEN, 91191, Gif-sur-Yvette, France^e Institut de Chimie et Procédés pour l'Énergie, l'Environnement et la Santé (ICPEES), UMR CNRS 7515, 67087 Strasbourg, France

† Electronic supplementary information (ESI) available: Details on the materials synthesis, their structural, thermal and electrochemical properties, theoretical calculations, device data are provided. 2D NMR studies and copies of NMR spectra are also included. See DOI: 10.1039/d0tc04650g

difficult task as removing the functional organic layers of a PhOLED stack leads to a dramatic decrease of the performance. Simplifying the PhOLED technology goes also through the use of high-efficiency universal materials which can efficiently host red (R), green (G) and blue (B) phosphors.

Some examples have shown that, with rational designs, the host material can perform the job of the numerous functional organic layers used in multi-layer structures. Thus, an ideal host material for RGB SL-PhOLEDs should fulfil several criteria: (i) a high triplet state energy $E_T > 2.7$ eV to confine the triplet excitons within phosphorescent guest, (ii) HOMO/LUMO energy levels well adapted to the electrode Fermi levels allowing efficient charge injection, (iii) good and well balanced mobilities of electrons and holes (ambipolar character) in order to compensate for the absence of electron/hole transporting layers,³⁹ and (iv) thermal and morphological stabilities to extend the lifetime of the devices. These four criteria can be fulfilled by the careful association of an electron-rich and an electron-deficient unit within a single molecule. However, while some examples of very high performance RGB multi-layer PhOLEDs have been recently described,^{5,18,40–45} RGB SL-PhOLEDs remain very rarely reported in literature.^{23,24}

Thus, in the present work, we consider the simplest EML of a SL-PhOLED only constituted of one host material and the phosphor. The literature also reports other strategies to reach high efficiency SL-PhOLEDs such as a host/co-host combination in the EML. Despite high EQE have been obtained, this strategy^{46–50} requires several molecules instead of only one in conventional SL-PhOLEDs. Other simplified device architectures have also been used in the literature, with one or two undoped regions of the host material on each side of the EML.^{51,52}

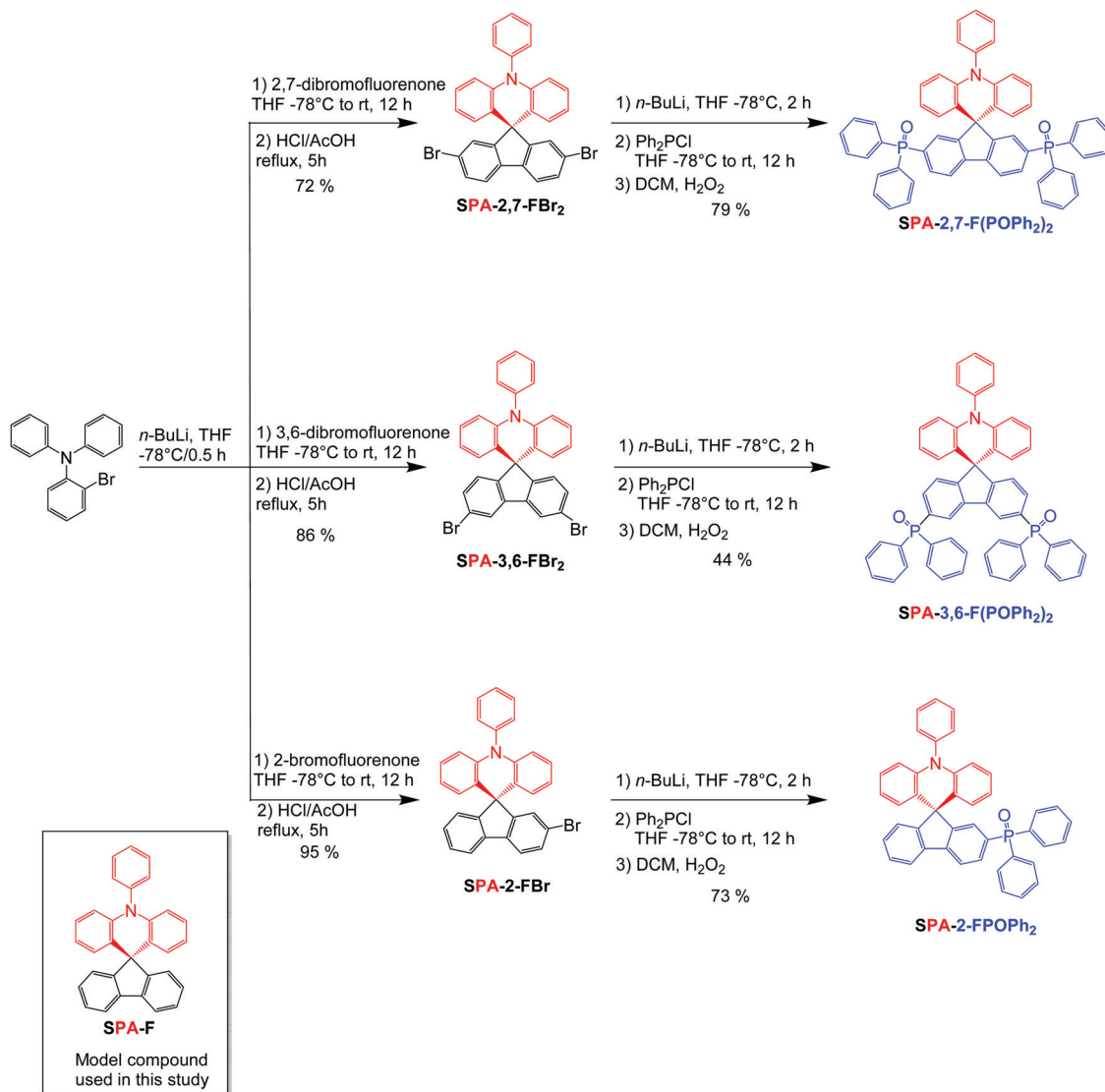
Recently, we have shown that phenylacridine/diphenylphosphine oxide association in **SPA-2,7-F(POPh₂)₂** (see structure in Scheme 1) is very efficient to reach high performance green and blue SL-PhOLEDs.³³ In the present work, we report a structure properties relationship study involving three bipolar hosts (**SPA-2,7-F(POPh₂)₂**,³³ **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂**) constructed on the association of the electron-rich phenylacridine and the electron-poor diphenylphosphine oxide. The position (C2 and C7 vs. C3 and C6) and the number (1 vs. 2) of diphenylphosphineoxide units on the fluorene backbone have been studied in detail to highlight the best combination in term of device performance. Such studies are the foundation of organic electronics to reach high performance devices. Finally, RGB SL-PhOLEDs have been fabricated and characterized using four different emitters (red: bis(2-methyldibenzo[*f,h*]quinoxaline)-(acetylacetonate)iridium(III)-Ir(MDQ)₂(acac), green: tris[2-phenylpyridinato-C²,N]iridium(III)-Ir(ppy)₃ and two blue emitters: bis(3,5-difluoro-2-(2-pyridyl)phenyl-(2-carboxypyridyl) iridium(III)-FIrpic and bis(2,4-difluorophenylpyridinato)-tetrakis(1-pyrazolyl)borate iridium(III)-FIR6, see characterization of iridium complexes in Table 2 and ESI†). It is important to mention that the blue emitter FIR6 investigated herein has been rarely used in literature (due to its high E_T and the resulting difficulty to be hosted). Some examples are found for multi-layer PhOLEDs^{40,53,54} but as

far as we know no example has been reported to date in SL-PhOLEDs using a single host (only one example exists and it involves a host:co-host system and displays low performance⁴⁷). In this work, we manage to reach a promising FIR6-based SL-PhOLED (with **SPA-2-FPOPh₂**) possessing an EQE of 9.1% and a low V_{on} below 3 V. This may allow the development of blue SL-PhOLEDs with emission wavelengths shorter than those of the sky blue emitter Firpic. On the other hand, with notably a high EQE of 18% with Firpic as sky blue emitter, **SPA-2,7-F(POPh₂)₂** displays the highest overall performance in the series and demonstrates that high performance SL-PhOLEDs can be reached for all the colours with the same host material. This translates the efficiency of the association of phenylacridine and diphenylphosphineoxide units in the EML of RGB SL-PhOLEDs and its potential for the future.

Synthesis

For potential industrial applications and to reduce the environmental footprint, the synthesis of a universal host material for a SL-PhOLED should (i) be short and high yielded, (ii) use inexpensive starting materials and (iii) avoid rare metal catalysts. The present target molecules have been synthesized following a versatile and efficient two-step route (Scheme 1). A lithium–bromine exchange was first performed on 2-bromophenyldiphenylamine followed by the trapping of the lithiated intermediate by the corresponding fluorenone, *i.e.* either 2,7-dibromofluorenone for **SPA-2,7-F(POPh₂)₂**, 3,6-dibromofluorenone for **SPA-3,6-F(POPh₂)₂** or 2-bromofluorenone for **SPA-2-FPOPh₂**. Spiro compounds **SPA-2,7-FBr₂**, **SPA-3,6-FBr₂** and **SPA-2-FBr** were then obtained by cyclization (in HCl/AcOH media) of the corresponding fluorenols (not isolated) in a high yield of 72%, 86% and 95%. Adding *n*-BuLi to these platforms lead to a lithium–halogen exchange reaction providing the corresponding lithiated intermediates, which were trapped with chlorodiphenylphosphine to provide the corresponding diphenylphosphine compounds (not isolated), further oxidized in the presence of H₂O₂ to give **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** or **SPA-2-FPOPh₂** with a yield of 79%, 44% and 73% respectively. Therefore, this synthetic approach is versatile, short, very efficient (overall yield of 57%, 38% and 69%), and low cost as it uses cheap starting materials (less than 0.5 € per g for 2-bromofluorenone, 2 € per g for 2,7-dibromofluorenone and 4 € per g for 3,6-dibromofluorenone, 8 € per g for 2-bromophenyldiphenylamine) and no palladium catalyst. It should be mentioned that **SPA-2,7-FBr₂**, **SPA-3,6-FBr₂** and **SPA-2-FBr** are appealing functional platforms, on which can be easily attached many different molecular fragments of interest for organic electronics.

In order to precisely study the impact of the incorporation of the electron-poor units within the three compounds, their properties will be compared to those of unsubstituted model compound spirophenylacridine-fluorene **SPA-F** (See molecular structure in the insert of Scheme 1).³³ Note that **SPA-2,7-F(POPh₂)₂** and **SPA-3,6-F(POPh₂)₂** are positional isomers, a key concept in organic chemistry, more and more used in the design of organic semi-conductors for electronics^{55,56}



Scheme 1 Synthesis of **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** and molecular structure of model compound **SPA-F**.

(for Organic Field-Effect Transistors,^{57–59} OLEDs,^{60–63} or Organic Photovoltaics⁶⁴).

The electrochemical properties of **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂**, **SPA-2-FPOPh₂** and the model compound **SPA-F** have been investigated by cyclic voltammetry (CV) in CH₂Cl₂ for oxidation and in DMF for reduction (Fig. 1, top); potentials are given *versus* a saturated calomel electrode (SCE).

In oxidation, **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** present three successive oxidation waves with maxima reported in Table 1 and detailed electrochemical studies presented in ESI†. For these three compounds, the first oxidation process is irreversible (at a sweep-rate of 100 mV s⁻¹) with a maximum close to 1.05 V (Fig. 1, top-right). The model compound **SPA-F**, without any phosphine oxide attached, displays a different behaviour with a first reversible oxidation wave at 1.00 V, being therefore shifted by 50 mV (Fig. 1, top-right, green line) compared to the three phosphine oxide compounds. This indicates that the phosphine oxide fragments have an influence

on the reversibility of the first oxidation wave and on its potential values. Indeed, despite the separation of the donor and the acceptor units by the spiro bridge, the oxidation of the phenylacridine is influenced by the nature of the electron poor bis(diphenylphosphineoxide)-fluorene unit and appears more difficult to oxidize than the model compound **SPA-F**. However, it is noteworthy to mention that the number and the position of the diphenylphosphine oxide units do not influenced significantly the oxidation of the phenylacridine unit as the **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** compounds are oxidized at similar potential values (*ca.* 1.05 V). In reduction, the results are different.

In reduction (see detailed electrochemical studies in ESI†), **SPA-2,7-F(POPh₂)₂** displays three successive reduction waves with maxima at -1.98, -2.50 and -2.79 V, whereas **SPA-3,6-F(POPh₂)₂** only displays two reduction waves with maxima at -2.40 and -2.83 V (Fig. 1, top-left). Contrary to **SPA-2,7-F(POPh₂)₂**, which presents a first reversible reduction process,

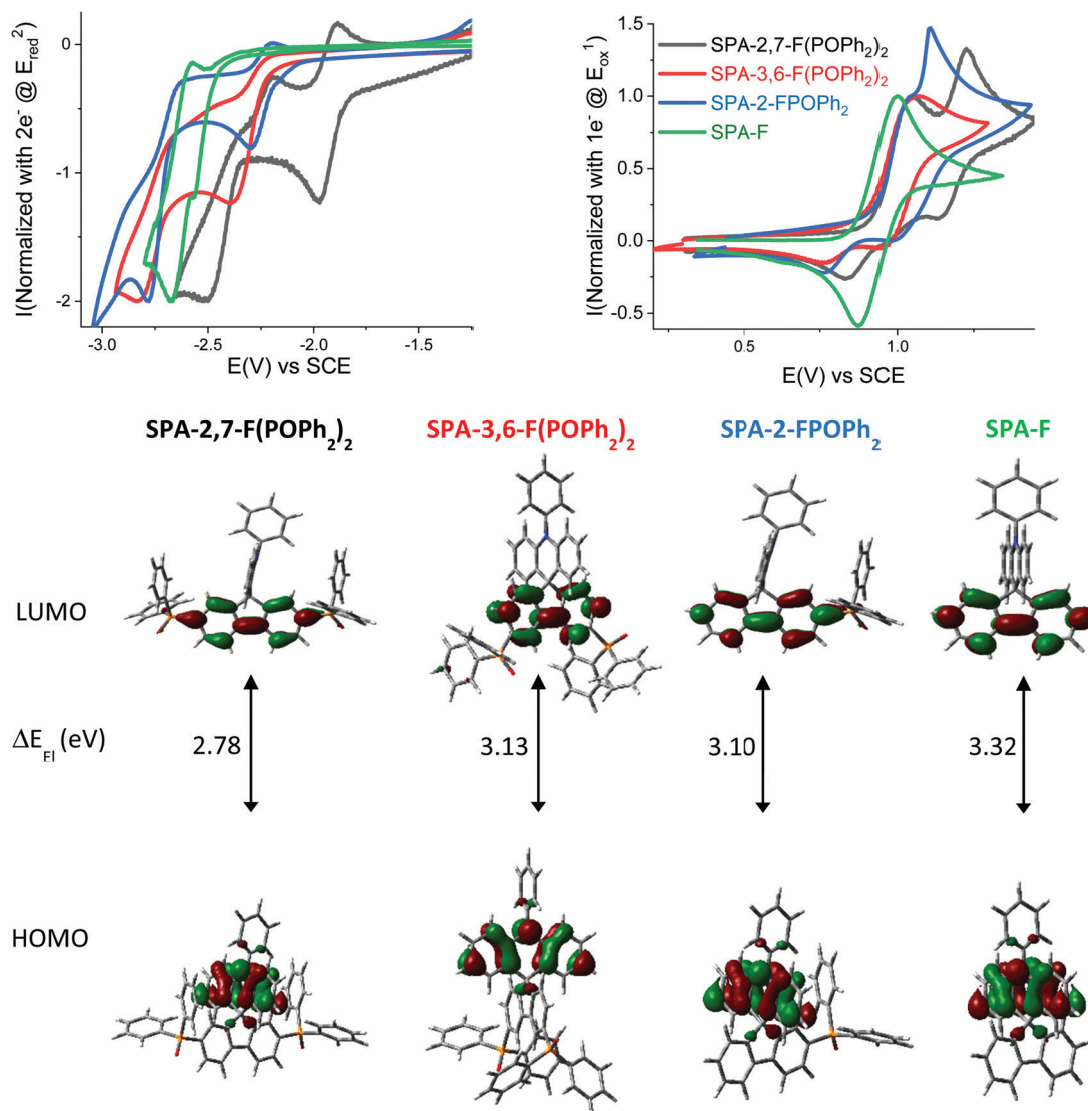


Fig. 1 Top: Normalized cyclic voltammograms of **SPA-2,7-F(POPh₂)₂** (black lines), **SPA-3,6-F(POPh₂)₂** (red lines) and **SPA-2-FPOPh₂** (blue lines) and **SPA-F** (green lines) in the cathodic (left, DMF + Bu₄NPF₆ 0.1 M) or the anodic (right, CH₂Cl₂ + Bu₄NPF₆ 0.2 M) range. Sweep-rate: 100 mV s⁻¹, platinum disk working electrode. Bottom: Frontier molecular orbitals LUMO and HOMO calculated by TD-DFT (b3lyp/6-311+G(d,p)), isovalue 0.04 [e Bohr^{-3/2}].

the two reduction processes of **SPA-3,6-F(POPh₂)₂** are irreversible, indicating a more reactive radical anion than for the latter. **SPA-2-FPOPh₂** also displays two successive reduction waves with maxima at -2.30 and -2.78 V, only the first one being reversible at 100 mV s⁻¹. Thus, the first reduction occurs at a different potential as a function of the number and the position of the diphenylphosphine oxide units: -1.98 V for **SPA-2,7-F(POPh₂)₂**, -2.30 V for **SPA-2-FPOPh₂** and -2.40 V for **SPA-3,6-F(POPh₂)₂**. Molecular modelling shows that this first electronic transfer is centred on the fluorene bearing the diphenylphosphine oxide units for the three compounds (see the electronic delocalization of the LUMO in Fig. 1, bottom).

The HOMO levels have been evaluated from the onset oxidation potential at -5.26 eV for model compound **SPA-F** and at -5.33 eV for both **SPA-2,7-F(POPh₂)₂** and **SPA-2-FPOPh₂** and at -5.31 eV for **SPA-3,6-F(POPh₂)₂**. Molecular modelling

shows that the HOMO of all compounds are exclusively spread out on the acridine moiety (Fig. 1, bottom) with levels calculated at -5.29 eV for **SPA-F**, -5.55 eV for **SPA-2,7-F(POPh₂)₂**, -5.43 eV for **SPA-2-FPOPh₂** and at -5.45 eV for **SPA-3,6-F(POPh₂)₂**. Thus, theoretical calculations confirm that, even if the donor part is localized for all four compounds on the phenylacridine moiety, the presence of POPh₂ fragments in **SPA-2,7-F(POPh₂)₂**, **SPA-2-FPOPh₂** and **SPA-3,6-F(POPh₂)₂** induces a decrease of the HOMO compared to **SPA-F**.

The LUMO levels obtained from the onset reduction potential are respectively evaluated at -2.55, -2.18, -2.23 and -1.94 eV for **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂**, **SPA-2-FPOPh₂** and **SPA-F**. The same trend is also observed through theoretical calculations: -1.88, -1.51, -1.59 and -1.20 eV for **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂**, **SPA-2-FPOPh₂** and **SPA-F**. The values of the LUMO levels of the three compounds

Table 1 Selected electronic and physical data

	SPA-2,7-F(POPh ₂) ₂	SPA-3,6-F(POPh ₂) ₂	SPA-2-FPOPh ₂	SPA-F
$\lambda_{\text{abs max}}^a$ [nm] ($\epsilon \times 10^4$)	323 (2.0); 310 (1.3); 295 (2.4); 283 (2.1); 274 (1.7)	316 (1.2); 304 (1.2); 273 (2.2)	315 (2.5); 310 (1.3); 295 (2.4); 283 (2.1); 274 (1.7)	309 (2.4)
$\lambda_{\text{em fluo}}^a$ [nm]	436	368	402	345
E_s^c [eV]	3.15	3.58	3.43	3.77
QY ^d	<0.01	<0.01	0.02	nd
$\lambda_{\text{em phospho}}^b$ [nm]	450	428	439	428
E_T [eV]	2.76	2.90	2.82	2.90
$E_s - E_T$ [eV]	0.39	0.68	0.61	0.87
τ_p [s] (λ_{em} [nm]) ^b	3.1 (450)	4.7 (428)	3.9 (439)	5.6 (428)
$E_{\text{ox}}^{\text{ef}}$ [V]	1.06, 1.23, 2.18	1.07, 2.25	1.04 (sh), 1.11, >2.2	1.00, 1.77, 2.20
$E_{\text{red}}^{\text{eg}}$ [V]	−1.98, −2.50, −2.79	−2.40, −2.83	−2.30, −2.78	−2.56, −2.67
HOMO ^h [eV]	−5.33	−5.31	−5.33	−5.26
LUMO ^h [eV]	−2.55	−2.18	−2.23	−1.94
ΔE_{el}^i [eV]	2.78	3.13	3.10	3.32
μ_{h}^j [cm ² V ^{−1} s ^{−1}]	8.2×10^{-6}	1.4×10^{-8}	1.9×10^{-7}	1×10^{-5}
μ_{e}^j [cm ² V ^{−1} s ^{−1}]	2×10^{-4}	3.1×10^{-6}	1.3×10^{-5}	—
T_d^k [°C]	474	411	381	286
T_g^l [°C]	143	165	118	90
T_c [°C]	218	—	191	141

^a In cyclohexane at RT, $\lambda_{\text{exc}} = 310$ nm. ^b In 2-MeTHF at 77 K, $\lambda_{\text{exc}} = 310$ nm. ^c From fluorescence onset at RT. ^d From first phosphorescence peak. ^e Vs. SCE. ^f In CH₂Cl₂. ^g In DMF. ^h From electrochemical data. ⁱ $\Delta E_{\text{el}} = |\text{HOMO} - \text{LUMO}|$. ^j Determined from SCLC devices analysis. ^k Determined by TGA. ^l Determined by DSC (2nd heating), nd: not determined.

are lower than the one of the model compound **SPA-F**, because of the strong electron-withdrawing character of diphenylphosphine oxide units directly linked to the fluorene core, where the LUMO is delocalized. The difference in terms of energy levels of the LUMO between **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** is assigned to both the number and the position of the phosphine oxide units. Thus, **SPA-2,7-F(POPh₂)₂** displays the lowest LUMO energy level in the series due to the position of the phosphine oxides at C2 and C7 of the fluorene unit. Indeed, these positions allow an electronic coupling (*para* position of the biphenyl linkage) between the substituent and the fluorene core as previously shown in literature with other fluorene based systems.^{55,60} The fluorene core is therefore strongly influenced by the inductive effect of the phenylphosphine oxides located at these positions. As only one phosphine oxide is attached at C2 of **SPA-2-FPOPh₂**, its LUMO energy is therefore slightly higher than that of **SPA-2,7-F(POPh₂)₂**. **SPA-3,6-F(POPh₂)₂** displays the highest LUMO energy, −2.18 eV, in the series as the phosphine oxides, located at C3/C6 (*meta* position of the biphenyl linkage), have a weaker electronic effect on the fluorene backbone than at C2/C7 (*para* position of the biphenyl linkage).⁶⁰

The electrochemical energy gap (difference between the HOMO and the LUMO energy level) of **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** are respectively evaluated at 2.78 eV, 3.13 eV and 3.10 eV, Table 1. These gaps are all contracted compared to that of **SPA-F**, (3.32 eV), however with a different magnitude, mainly due to the different LUMO energy levels of the three compounds. Thus, **SPA-2,7-F(POPh₂)₂** displays the shortest gap, followed by **SPA-2-FPOPh₂** and by **SPA-3,6-F(POPh₂)₂**, showing that the gap can be easily controlled by the number and the position of the phosphine oxide units borne by the fluorene backbone. This gap contraction is a central point in the design of host materials for SL-PhOLEDs as

an excellent injection of hole and electron is mandatory (adjustment of HOMO and LUMO energy levels). This feature will be discussed below in the PhOLED part.

In UV-vis absorption spectroscopy (Fig. 2, top-left), model compound **SPA-F** displays a small band at 309 nm and a long tail until 350 nm. Thanks to Time Dependent Density Functional Theory (TD-DFT) calculations (Fig. 3), the small band can be attributed to two transitions: HOMO to L+5 both localized on the acridine part ($\lambda_{\text{th}} = 305$ nm) and a H−1 to LUMO both localized on the fluorene ($\lambda_{\text{th}} = 292$ nm). Adding one phosphine unit at C2 of the fluorene backbone in **SPA-2-FPOPh₂** induces a red shift of the main band, recorded at 315 nm. The shift of this band confirms that the phosphorus atom is not fully insulating in such a system and that the whole π -conjugation is extended. This band can be attributed to two transitions both simulated at 304 nm and implying the same major contributions: a charge transfer one from the HOMO on the acridine to the π^* L+5 localized on the phosphine oxide fragment and a second one H−1 → LUMO, with both orbitals localized on the fluorene. Adding a second phosphine oxide unit in **SPA-2,7-F(POPh₂)₂** leads to a further 8 nm red shift, with a band centred at 323 nm. This band can be assigned to a transition modeled at 315 nm with both orbitals centred on the fluorene (H−1 → LUMO). One can note that the shape of the absorption spectra of **SPA-2-FPOPh₂** and **SPA-2,7-F(POPh₂)₂** is very similar and a characteristic of the substitution at C2/C7. The absorption spectrum of **SPA-3,6-F(POPh₂)₂** is less defined and displays lower absorption coefficients for the low energy bands. The main band is recorded at 316 nm indicating that shifting the phosphine oxides from the C2/C7 to the C3/C6 position leads to a 7 nm blue shift. TD-DFT indicates that this band is due to three π - π^* transitions: two are combinations of contributions from orbitals all localized on the fluorene, H−1 → LUMO and H−1 → L+1 ($\lambda_{\text{th}} = 290$ and 299 nm), and the last

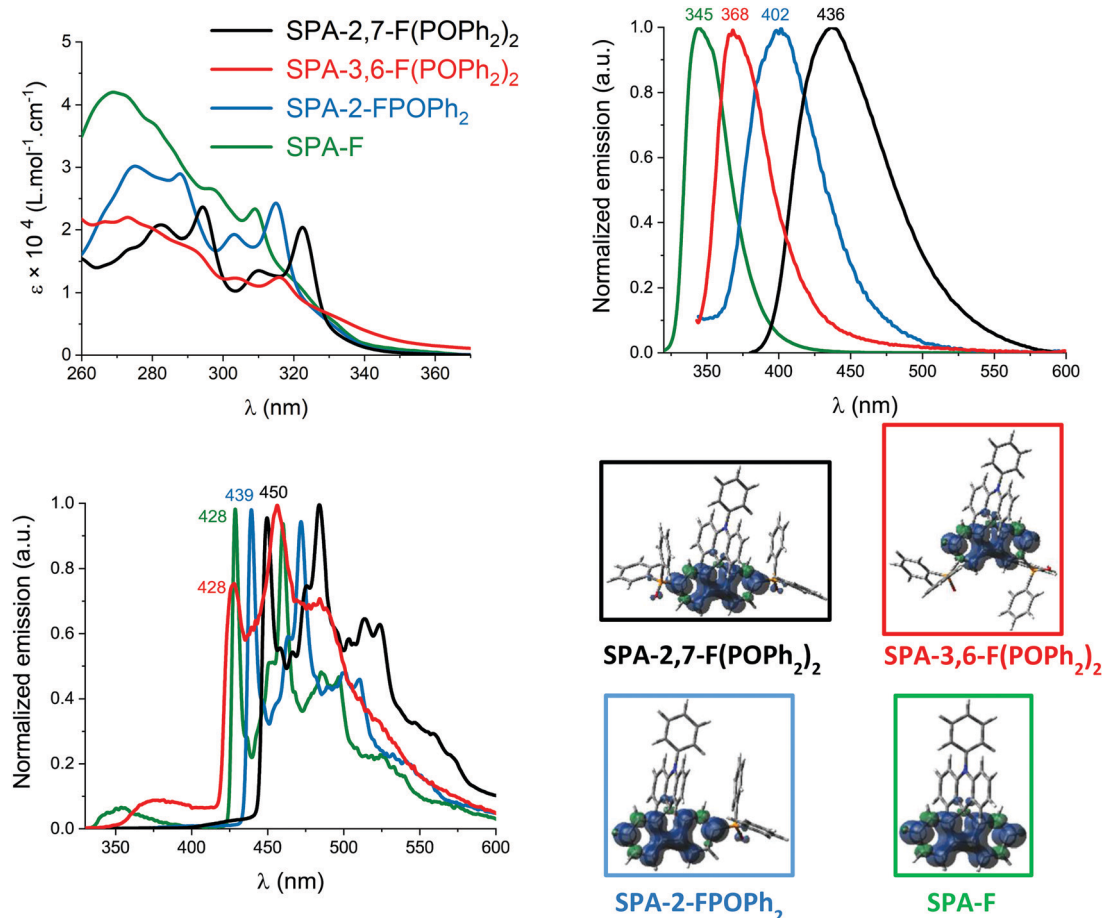


Fig. 2 UV-vis absorption spectra in cyclohexane (top-left). Normalized emission spectra at room temperature in cyclohexane, $\lambda_{\text{exc}} = 310 \text{ nm}$ (top-right). Emission spectra at 77 K in 2-MeTHF ($\lambda_{\text{exc}} = 310 \text{ nm}$) normalized at the phosphorescence maxima (bottom-left). Triplet spin density distribution (TD-DFT, b3lyp/6-311+g(d,p), isovalue 0.002, bottom right) of **SPA-2,7-F(POPh₂)₂** (black lines) **SPA-3,6-F(POPh₂)₂** (red lines), **SPA-2-FPOPh₂** (blue lines) and **SPA-F** (green lines).

transition implies orbitals on the acridine part, HOMO \rightarrow L+8 ($\lambda_{\text{th}} = 314 \text{ nm}$). The important tail observed for **SPA-3,6-F(POPh₂)₂** is due to several transitions at lower energies implying contributions from HOMO (π orbital on the acridine) to π^* orbitals such as L+2, L+3 (both localized on the fluorene and POPh₂ fragments), or L+5 (localized on the phenylacridine).

Note that the TD-DFT calculations of the four compounds show that the first excited state corresponds to a forbidden HOMO–LUMO transition, not detectable experimentally, Fig. 3. This is due to the spatial separation of HOMO and LUMO levels (HOMO localized on the acridine core and LUMO on the substituted fluorene core, Fig. 3) leading to a through-space forbidden transition.⁶⁵ This feature is caused by the insulating spiro bridge (despite a weak coupling exists as shown above in the electrochemical analyses) and indicates that the electronic coupling between the electron-rich unit and the electron-poor units is efficiently restrained, which is a key point in the present design to maintain a high triplet state energy level E_{T} , as exposed below.

Two features can be concluded. First, the red shift observed for **SPA-3,6-F(POPh₂)₂** compared to **SPA-F** shows that despite a

meta linkage is involved in **SPA-3,6-F(POPh₂)₂**, electronic coupling exists between the fluorene and the phosphine oxide units. This feature has recently been approached in literature for *meta* linked spirobifluorene based materials.^{10,60} Second, it is interesting to note that the effect of one phosphine oxide at C2 is similar in term of absorption wavelength to that of two at C3/C6 (the two molecules have almost identical λ_{max}). This shows how the absorption properties can be tuned by the number and the position of the substituents.

In fluorescence spectroscopy (cyclohexane, Fig. 2-top right), the spectra are unresolved and we note a gradual red shift of their maxima from **SPA-F** (345 nm), **SPA-3,6-F(POPh₂)₂** (368 nm), **SPA-2-FPOPh₂** (402 nm), and to **SPA-2,7-F(POPh₂)₂** (436 nm). Thus, adding two phosphine oxides at C3/C6 in **SPA-3,6-F(POPh₂)₂** leads to a significant red shift of *ca.* 20 nm, showing again the electronic coupling between the fluorene and the phosphine oxides. This red shift is even more pronounced, *ca.* 60 nm, for **SPA-2-FPOPh₂** due to the substitution at C2 (*para* position). The double substitution at C2/C7 in **SPA-2,7-F(POPh₂)₂** leads to an impressive 90 nm shift. Also, from **SPA-F** to **SPA-2,7-F(POPh₂)₂** the fluorescence band

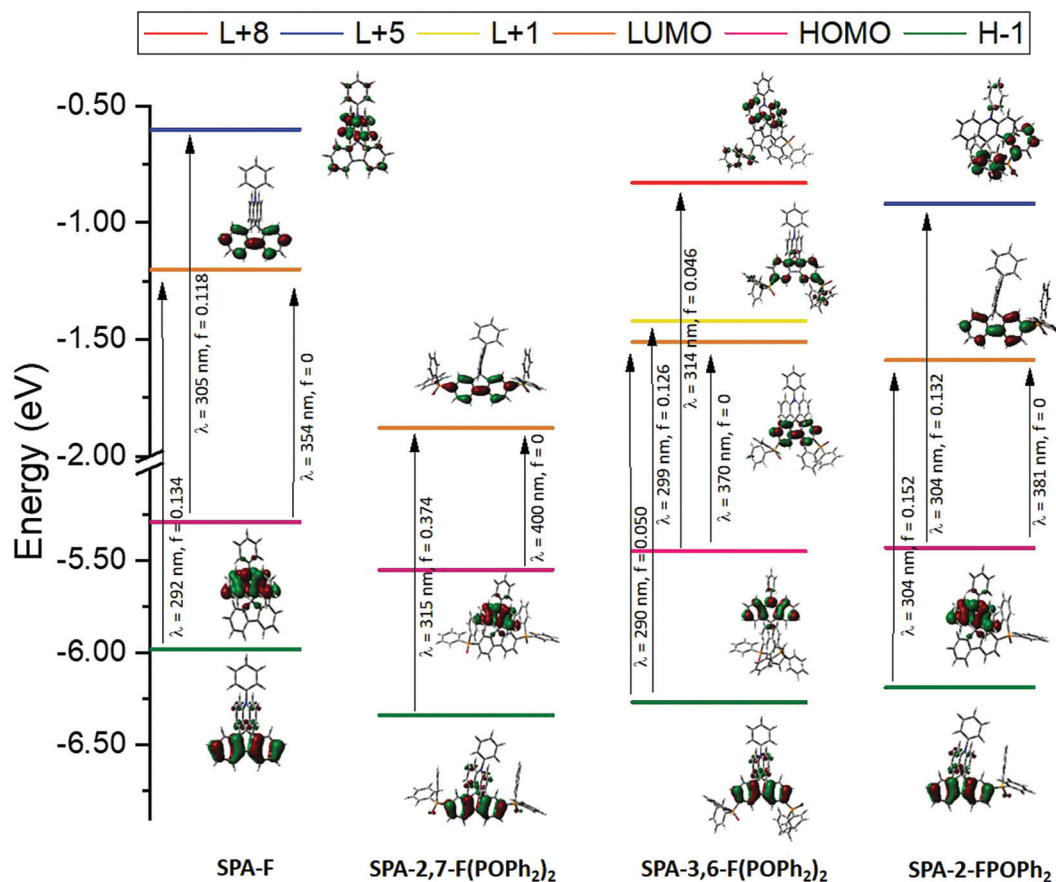


Fig. 3 Representation of the energy levels and the main molecular orbitals involved in the electronic transitions of **SPA-F**, **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** obtained by TD-DFT B3LYP and the 6-311+G(d,p) basis set on the geometry of *S*₀, shown with an isovalue of 0.04 [*e* Bohr⁻³]^{1/2} (for clarity purpose, only the main contribution for each transition is shown, details provided in ESI†).

becomes larger and larger. This trend is in accordance with that highlighted in absorption but the red shifts from **SPA-F** to **SPA-2,7-F(POPh₂)₂** are larger due to a charge transfer character more and more important from **SPA-F** to **SPA-2,7-F(POPh₂)₂**.

Finally, the three investigated compounds are very bad emitters with quantum yields below 0.02 (Table 1). This is in accordance with the spatial separation of HOMO and LUMO (HOMO/LUMO transition presents an oscillator strength of *ca.* 0, Fig. 3). This characteristic is usually found in host materials for phosphors.^{65–67}

The phosphorescent contributions evaluated thanks to the emission spectra at 77 K in 2-MeTHF are well resolved, very similar in shape but different in terms of wavelengths. The lowest *E*_T in the series is logically found for **SPA-2,7-F(POPh₂)₂**, 2.76 eV, which displays a first phosphorescence contribution at 450 nm (Fig. 2, bottom-left). Molecular modelling indicates that the *E*_T of 2,7-(POPh₂)₂-SPA is fully governed by the diphenylphosphineoxide-fluorene fragment as visualized by the triplet spin density found on the fluorene core and slightly on the phenyl units of the phosphine oxide (Fig. 2, bottom-right). Removing one phosphine oxide in **SPA-2-FPOPh₂** increases the *E*_T by 0.06 eV (2.82 eV, λ = 439 nm) and concentrates the triplet spin density on the fluorene core. The highest

*E*_T in the series is found for **SPA-3,6-F(POPh₂)₂**, which displays a very high *E*_T of 2.90 eV (λ = 428 nm) due to its double *meta* connection. This value is identical to that of model compound **SPA-F** (2.90 eV), meaning that the effect of the fluorene substitution is completely cancelled in the triplet state (note that the triplet spin density is exclusively spread out on the fluorene core, Fig. 2, bottom-right). This result is different to that observed for *S*₁ but in accordance with previous reports on the different trend observed between *S*₁ and *T*₁ states.^{56,60,68} Indeed, it has been previously shown that *meta* linkages of a fluorene core lead to a red shift of both absorption and fluorescence but do not modify the first phosphorescence contribution. This interesting design strategy seems to be easily adapted to the present fluorene/phosphine oxide systems. For all compounds, the emission from *T*₁ state is confirmed by the very long lifetime measured at 77 K for these four compounds (τ = 3.1, 3.9, 4.7 and 5.6 s for **SPA-2,7-F(POPh₂)₂**, **SPA-2-FPOPh₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-F** respectively, Table 1).

It should finally be stressed out that, at 77 K, a very weak fluorescence contribution is observed for the three compounds **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-F** and almost none for **SPA-2-FPOPh₂**. This is a different behaviour than that observed for many other host materials, in which the

fluorescence is very intense at 77 K.^{4,69} This feature is connected to the very low fluorescence quantum yield measured at room temperature for all compounds. Indeed, the intersystem crossing between S_1 and T_1 is favoured, and leads at 77 K to an intense phosphorescence contribution and a weak fluorescence contribution.^{70,71} Note that, in the case of **SPA-2,7-F(POPh₂)₂**, the S_1 and T_1 states are the closest in the series (<0.4 eV) and the fluorescence and phosphorescence contributions are overlapped (S_1 energies have been evaluated from the onset of the emission spectra at room temperature and T_1 energies have been evaluated from the peak maximum at 77 K as often found in literature⁷²). This S_1/T_1 gap increases by shifting the phosphine oxide units from C2/C7 to C3/C6 (0.7 eV for **SPA-3,6-F(POPh₂)₂**) or by removing one phosphine oxide unit (0.6 eV for **SPA-2-FPOPh₂**) or two (0.9 eV for **SPA-F**). In fact, both S_1 and T_1 states decrease upon conjugation expansion, with a significantly more important effect on the S_1 states compared to T_1 states. This is why, in the case of the highly conjugated **SPA-2,7-F(POPh₂)₂**, the S_1 state decreases close to the T_1 state.

The thermal properties have been studied by thermogravimetric analyses (TGA, see ESI†) and differential scanning calorimetry (DSC, Fig. 4). Due to the presence of the rigid spiro bridge and bulky diphenylphosphine oxides, **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** display higher decomposition temperature T_d (5% mass loss) than unsubstituted **SPA-F** ($T_d = 286$ °C). The highest T_d is recorded for **SPA-2,7-F(POPh₂)₂** at ca. 474 °C (see ESI†). The positive influence of the diphenylphosphine oxide units on the thermal properties is confirmed by DSC. Thus, during the 2nd heating run (between 20 and 350 °C), the glass transition temperature T_g increases from **SPA-F** to **SPA-2-FPOPh₂** when adding one phosphine oxide (90 vs. 118 °C, Fig. 4). The T_g are even much increased when two phosphine oxides are present within the molecular structure, 143 °C for **SPA-2,7-F(POPh₂)₂** and 165 °C for **SPA-3,6-F(POPh₂)₂**. One can hence note that the positions C3/C6 provide the material with the highest T_g .

One can also note that both **SPA-2,7-F(POPh₂)₂** and **SPA-2-FPOPh₂** respectively present a crystallization temperature T_c at

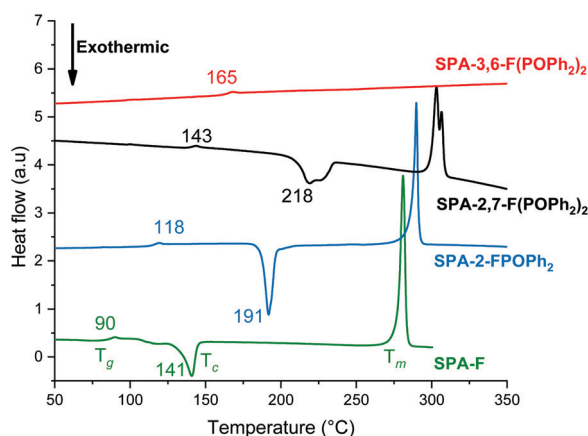


Fig. 4 DSC traces (2nd heating) of **SPA-2,7-F(POPh₂)₂** (black line), **SPA-3,6-F(POPh₂)₂** (red line), **SPA-2-FPOPh₂** (blue line) and **SPA-F** (green line).

ca. 218 and 191 °C (during the 2nd heating cycle), also observed for **SPA-F** but at a much lower temperature, i.e. 141 °C (Fig. 4). Thus, incorporating one or two diphenylphosphine oxide units in C2/C7 allows to significantly increase the T_c . The substitution at *meta* positions (C3/C6) of **SPA-3,6-F(POPh₂)₂** leads to a different result as neither T_c nor T_m (melting) transitions are observed during the 2nd heating cycle, of great interest for further OLED applications.

The device architecture simplification, by removing interfacial layers, requires that the host matrix can efficiently act simultaneously as hole and electron charge carrier. In order to allow recombination of opposite charges within the EML, it is moreover necessary to have a good balance between hole and electron carrier mobilities. Otherwise, excitons are created near an electrode, which increases the quenching rate due to non-radiative recombination. In this respect, the charge transport property characterization is mandatory in order to rationalize the structure – device efficiency relationship in these host materials for PhOLED applications. As PhOLEDs are vertical devices, space charge limited current (SCLC) devices appear to be the most appropriate devices to probe the charge carrier mobilities. Indeed, by elaborating hole-only and electron-only devices, one can access the out-of-plane hole and electron mobilities respectively (see ESI† for composition and elaboration details).

The hole/electron mobility (μ_h/μ_e) of **SPA-2,7-F(POPh₂)₂**, **SPA-2-FPOPh₂**, and **SPA-3,6-F(POPh₂)₂** have been estimated to be $8.2 \times 10^{-6}/2.0 \times 10^{-4}$ cm² V⁻¹ s⁻¹, $1.9 \times 10^{-7}/1.3 \times 10^{-5}$ cm² V⁻¹ s⁻¹ and $1.4 \times 10^{-8}/3.1 \times 10^{-6}$ cm² V⁻¹ s⁻¹, respectively (Fig. 5). One can first interestingly note that the electron mobility is higher than that of hole for the three hosts studied herein. This is a key point as it allows to balance the hole transporting nature of the iridium complexes (FIrpic,⁷³ Ir(ppy)₃,⁷⁴ no data found in literature for Ir(MDQ)₂(acac)). It is always awkward to explain trends in mobility values from a series of molecules. However, one can note that **SPA-2,7-F(POPh₂)₂**, which is the molecule with the larger conjugation (compared to **SPA-3,6-F(POPh₂)₂**), the greatest symmetry (compared to **SPA-2-FPOPh₂**), and the highest electron affinity (i.e. lowest LUMO level), exhibits the highest electron and hole mobilities. With a single phosphine oxide unit, **SPA-2-FPOPh₂** exhibits both structural asymmetry and lower electron affinity. As a consequence, both mobilities decrease by more than one order of magnitude. Finally, **SPA-3,6-F(POPh₂)₂** has the lowest charge carrier mobilities. This is in good agreement with the amorphous nature of this derivative, as shown by DSC analysis.

In addition, if we consider the mobility balance between electrons and holes (μ_e/μ_h) in each molecule, calculated at ca. 24, 68 and 221 for **SPA-2,7-F(POPh₂)₂**, **SPA-2-FPOPh₂**, and **SPA-3,6-F(POPh₂)₂**, respectively, **SPA-2,7-F(POPh₂)₂** appears clearly as the most suitable candidate to transport charges into a PhOLED device.

The different molecules were finally incorporated as host in red, green and blue SL-PhOLEDs using as emitter either Ir(MDQ)₂(acac) for red emission, Ir(ppy)₃ for green emission, and FIrpic or FIr6 for blue emission (Tables 2, 3 and Fig. 6, 7).

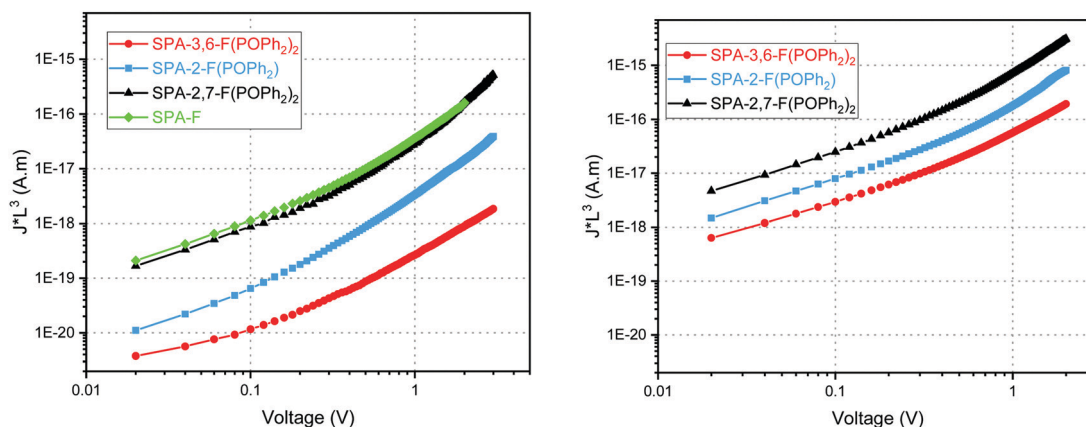


Fig. 5 Thickness-scaled current voltage characteristics of SPA-2,7-F(POPh₂)₂ (black lines), SPA-3,6-F(POPh₂)₂ (red lines), SPA-2-FPOPh₂ (blue lines) and SPA-F (green lines) hole- (left) and electron-only (right) SCLC devices.

The SL-PhOLED architecture is the following: ITO/PEDOT:PSS (40 nm)/EML (host + guest 10% wt) (100 nm)/LiF (1.2 nm)/Al (100 nm) with ITO/PEDOT:PSS as anode and LiF/Al as cathode.

First, the four iridium complexes have been studied in strictly identical conditions in order to precisely determine their electronic properties and particularly their HOMO and LUMO energy levels and E_T (Table 2). It is important to mention that many data can be found in literature for these phosphors but recorded in different experimental conditions (Firpic,⁷³ Ir(ppy)₃,^{75,76} Fir6,⁷⁷ Ir(MDQ)₂(acac)^{78,79}). This heterogeneity renders the comparison with host materials and between them difficult. In the case of SL-PhOLED, studying both host materials and phosphors in identical experimental conditions appear particularly important.

Thus, in 2-MeTHF at room temperature, the E_T of Ir(MDQ)₂(acac), Ir(ppy)₃, Firpic and Fir6 have been evaluated at 2.02, 2.43, 2.67 and 2.72 eV respectively from the maximum of first emission peak. At 77 K in a frozen 2-MeTHF matrix (in the same conditions than those used to measure the E_T of the hosts), there is a blue shift of the emission spectra (Table 2) due to the decrease of molecular motions and the E_T are therefore slightly increased at 2.08, 2.51, 2.72 and 2.76 eV. In thin films, dispersed into the host materials studied herein (corresponding to the EML of the devices studied below), the spectra are red shifted and the E_T are measured at 1.97, 2.40, 2.63 and 2.70 eV (note that the spectra of each complex are

independent of the matrix used, see ESI†). In these conditions, the thin-film phosphorescent spectra appear to be identical to the electroluminescent (EL) spectra presented below, showing the similitude of the optical processes involved.

The HOMO/LUMO of Ir(MDQ)₂(acac), Ir(ppy)₃, Firpic and Fir6 have also been evaluated by electrochemical analyses in solution (in CH₂Cl₂ + Bu₄NPF₆ 0.2 M, see CV in ESI†) at −5.26/−2.91, −4.97/−2.19, −5.55/−2.52, −5.66/−2.32 eV. These data are particularly interesting to interpret the device performance presented below.

First, the phosphor Ir(MDQ)₂(acac) (HOMO = −5.26/LUMO = −2.91 eV, E_T = 1.97 eV dispersed in the present hosts, Table 2) has been used as red-emitting dopant in SL-PhOLEDs. The best performance has been obtained with SPA-2,7-F(POPh₂)₂ as host reaching a maximum EQE of 8.7%. The best device reaches a maximum luminance L of 6843 cd m^{−2} at 170 mA cm^{−2} showing a good stability at high current density. The two other hosts SPA-3,6-F(POPh₂)₂ and SPA-2-FPOPh₂, studied in strictly identical experimental conditions display lower performance with low EQE of ca. 5%, Table 3. This can be related to the strong difference observed in terms of charge carrier mobilities as exposed above. Indeed, the hole and electron mobilities of SPA-2,7-F(POPh₂)₂ are not only significantly higher than those of both SPA-3,6-F(POPh₂)₂ and SPA-2-FPOPh₂ but also more balanced. When the charge transport is not balanced, the recombination zone is located close to the metal electrodes,

Table 2 Selected electronic and physical data of the four iridium complexes investigated

	Ir(MDQ) ₂ (acac)	Ir(ppy) ₃	Firpic	Fir6
λ_{em}^a [nm] rt/77 K/film ^b	613/597/630	511/493/517	465/456/471	456/449/460
E_T^{ac} [eV] RT/77 K/film ^b	2.02/2.08/1.97	2.43/2.51/2.40	2.67/2.72/2.63	2.72/2.76/2.70
E_{ox}^{de} [V]	0.96, 1.67	0.69, 1.74	1.28, 1.80	1.4 (sh), 1.56
E_{red}^{de} [V]	−1.68 (sh)	No distinct wave	−2.04	−2.40 (sh)
HOMO ^f [eV]	−5.26	−4.97	−5.55	−5.66
LUMO ^f [eV]	−2.91	−2.19	−2.52	−2.32
ΔE_{el}^g [eV]	2.35	2.78	3.03	3.34

^a In 2-MeTHF, λ_{exc} = 310 nm. ^b Dispersed in the host materials, either in SPA-2,7-F(POPh₂)₂, SPA-3,6-F(POPh₂)₂ or SPA-2-FPOPh₂. ^c From first phosphorescence peak. ^d Onset potential vs. SCE. ^e In CH₂Cl₂. ^f From electrochemical data (onset oxidation or reduction potentials).

^g ΔE_{el} = |HOMO–LUMO|.

Table 3 Best SL-PhOLEDs performance using **SPA-2,7-F(POPh₂)₂**, **SPA-3,6-F(POPh₂)₂** and **SPA-2-FPOPh₂** as host material. Device structure: ITO/PEDOT:PSS (40 nm)/host + dopant (100 nm)/LiF (1.2 nm)/Al (100 nm). % of the phosphor used: 10% in mass

V_{on} (V)	EQE (%)		CE (cd A ⁻¹)	PE (lm W ⁻¹)	L (cd m ⁻²)	EQE (%)	CE (cd A ⁻¹)	PE (lm W ⁻¹)	L (cd m ⁻²)	CIE coordinates (x; y)
	At 10 mA cm ⁻²					Max (at J (mA cm ⁻²))				At 10 mA cm ⁻²
Red PhOLEDs (10% Ir(MDQ) ₂ (acac))										
SPA-2,7-F(POPh ₂) ₂	2.8	6.0	6.2	2.0	1501	8.7 (0.03)	9.1 (0.03)	7.0 (0.03)	6843 (170)	0.64; 0.36
SPA-3,6-F(POPh ₂) ₂	3.6	4.4	4.4	1.3	1007	5.3 (0.04)	5.4 (0.04)	3.0 (0.04)	2973 (120)	0.64; 0.36
SPA-2-FPOPh ₂	3.2	2.6	3.2	1.1	673	4.5 (0.04)	5.5 (0.04)	3.5 (0.04)	151 (220)	0.63; 0.37
Green PhOLEDs (10% Ir(ppy) ₃)										
SPA-2,7-F(POPh ₂) ₂	2.3	11.0	37.8	18.2	9946	16.4 (0.04)	56.3 (0.04)	53.6 (0.04)	38970 (180)	0.31; 0.63
SPA-3,6-F(POPh ₂) ₂	2.7	9.4	35.3	13.8	8099	13.9 (0.03)	52.0 (0.03)	38.9 (0.03)	18610 (150)	0.31; 0.63
SPA-2-FPOPh ₂	2.7	7.5	25.6	13.4	6386	10.4 (0.02)	35.6 (0.02)	32.9 (0.02)	15490 (90)	0.31; 0.63
Sky Blue PhOLEDs (10% FIrpic)										
SPA-2,7-F(POPh ₂) ₂	2.5	12.5	27.3	14.5	5276	18.0 (0.04)	39.0 (0.04)	38.4 (0.04)	8030 (80)	0.15; 0.37
SPA-3,6-F(POPh ₂) ₂	3.5	6.5	12.3	4.0	2013	6.5 (6.8)	12.3 (6.8)	4.2 (6.8)	2540 (50)	0.16; 0.38
SPA-2-FPOPh ₂	2.8	8.0	16.2	6.3	3896	8.6 (0.24)	17.3 (0.24)	10.5 (0.24)	9578 (100)	0.15; 0.37
Blue PhOLEDs (10% FIr6)										
SPA-2,7-F(POPh ₂) ₂	2.8	5.4	8.6	4.3	2220	6.5 (27.7)	10.4 (27.7)	4.5 (27.7)	2687 (50)	0.16; 0.33
SPA-3,6-F(POPh ₂) ₂	4.6	5.1	11.5	4.2	604	6.5 (1.2)	14.7 (1.2)	6.8 (1.2)	661 (40)	0.16; 0.32
SPA-2-FPOPh ₂	2.9	9.0	15.9	7.9	3518	9.1 (14.6)	16.2 (14.6)	7.7 (14.6)	4952 (60)	0.15; 0.30

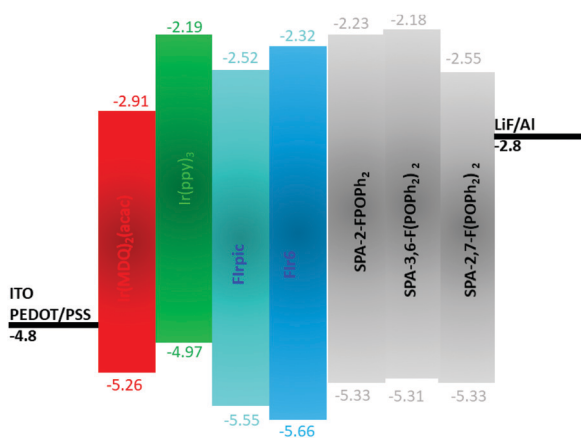


Fig. 6 Schematic energy diagrams of the different components used in the EML of the present SL-PhOLEDs.

thereby causing exciton quenching by the electrodes and reducing the device efficiency. This feature is surely at the origin of the higher device performance observed for red SL-PhOLEDs using **SPA-2,7-F(POPh₂)₂** as host. This will be confirmed with the other dopants presented below. On the other hand, the very low LUMO level of Ir(MDQ)₂(acac), -2.91 eV, is also involved in the moderate performance observed for the three hosts, the host possessing the lowest LUMO, *i.e.* **SPA-2,7-F(POPh₂)₂**/LUMO = -2.55 eV leading to the highest performance. The three devices also display a different threshold voltage (V_{on}): 2.8 V for **SPA-2,7-F(POPh₂)₂**, 3.2 V for **SPA-2-FPOPh₂** and 3.6 V for **SPA-3,6-F(POPh₂)₂**, translating (i) the different charge injections within the devices. This is in accordance with the different energy gaps observed for the three compounds (2.78 eV for **SPA-2,7-F(POPh₂)₂**, 3.10 eV for **SPA-2-FPOPh₂** and 3.13 eV for **SPA-3,6-F(POPh₂)₂**) the lowest gap leading to the lowest V_{on} and herein to the highest EQE. The different V_{on} can

also be related to the different μ_e/μ_h ratio of the hosts, the highest μ_e/μ_h leading to the highest V_{on} in the case of **SPA-3,6-F(POPh₂)₂**, and the lowest μ_e/μ_h leading to the lowest V_{on} in the case of **SPA-2,7-F(POPh₂)₂**.

The green SL-PhOLEDs have been then studied in a similar way using the classical green emitter Ir(ppy)₃ (HOMO = -4.97 eV, LUMO = -2.19 eV, E_T = 2.40 eV dispersed in the present hosts, Table 2). Usually, green-emitting SL-PhOLEDs display the highest performances (compare to blue and red) as green phosphors are usually the easiest to host in a PhOLED. The difference in terms of performances between the three hosts is less marked than for the red phosphor. Indeed, if the highest performance has been again reached with **SPA-2,7-F(POPh₂)₂**, the two other host materials also display interesting performances. The lowest efficiency is recorded for **SPA-2-FPOPh₂**, which displays a maximal EQE of 10.4% , and corresponding CE of 35.6 cd A⁻¹ and PE of 32.9 lm W⁻¹ at 0.02 mA cm⁻² (Table 3). The performances are increased with **SPA-3,6-F(POPh₂)₂** as a high maximal EQE of 13.9% and corresponding CE of 52.0 cd A⁻¹ and PE of 38.9 lm W⁻¹ (at 0.03 mA cm⁻²) are recorded. The best performance is finally obtained with **SPA-2,7-F(POPh₂)₂** as host with a maximal EQE of 16.4% , and corresponding CE of 56.3 cd A⁻¹ and PE of 53.6 lm W⁻¹ at 0.04 mA cm⁻² (Table 3). A maximum luminance of 38970 cd m⁻² at 180 mA cm⁻² is reached, translating a high performance and a good stability at high current density. In accordance with the data exposed above for the red phosphor, **SPA-2,7-F(POPh₂)₂** displays the lowest V_{on} in the series, 2.3 V due to its most contracted gap and lowest μ_e/μ_h in the series. During the writing of this manuscript, Isobe and coworkers have reported green SL-PhOLEDs with a higher EQE, overpassing 20% .³⁴

Blue phosphors are the most difficult to host and blue-emitting PhOLEDs remain the weakest link of the technology.^{5,18,42,68,80–82} In this work, two blue phosphors have been used: the classical sky blue emitter FIrpic (HOMO = -5.55 eV/LUMO = -2.52 eV, E_T = 2.63 eV dispersed in the

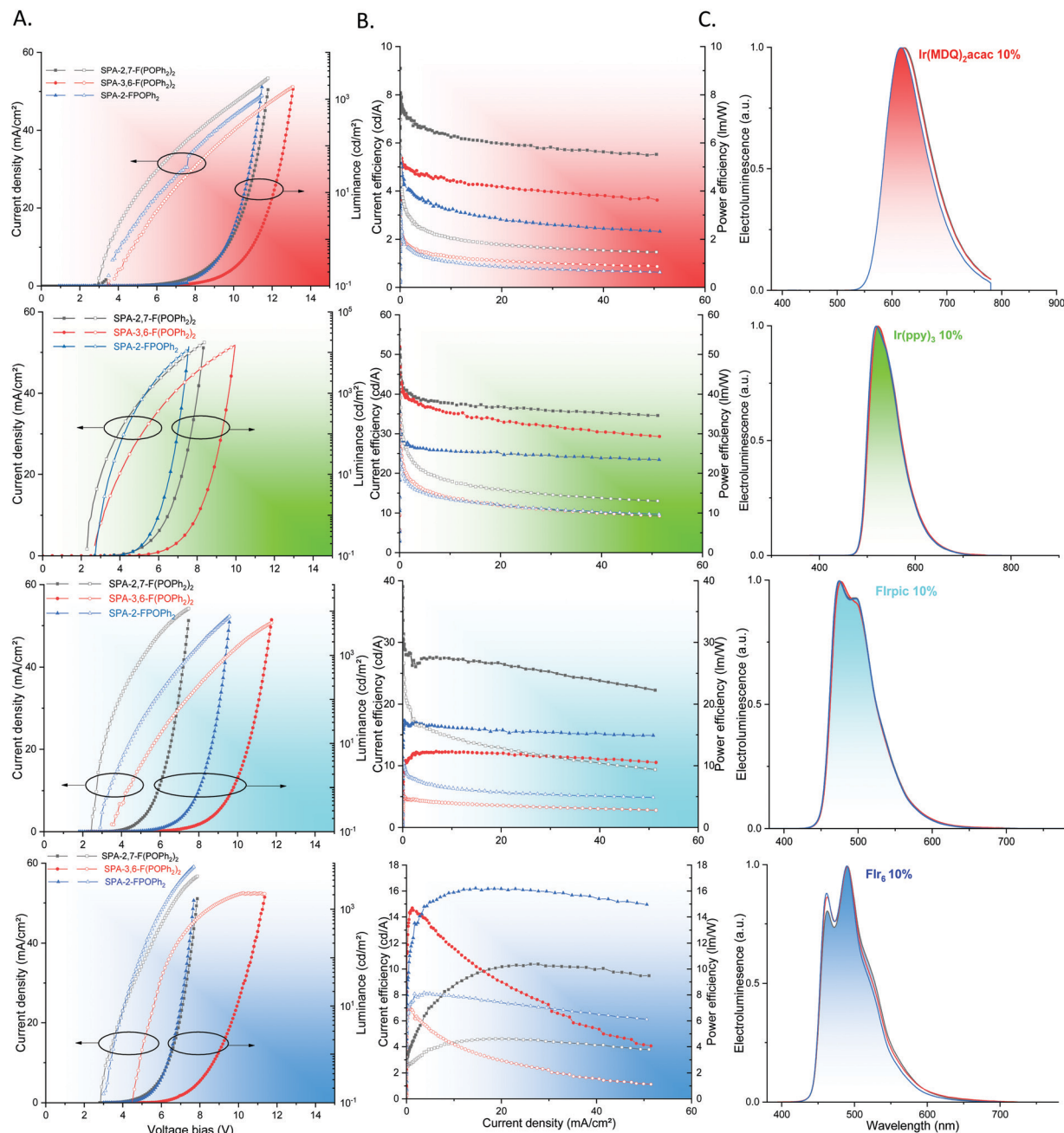


Fig. 7 SL-PhOLEDs characteristics using as a host either **SPA-2,7-F(POPh)₂** (black lines), **SPA-3,6-F(POPh)₂** (red lines) or **SPA-2-FPOPh₂** (blue lines). (A) Current density (mA cm^{-2}) and luminance (cd m^{-2}) as a function of the bias voltage; (B) current efficiency (cd A^{-1} , filled symbols) and power efficiency (lm W^{-1} , empty symbol) as a function of the current density (mA cm^{-2}) and (C) normalized EL spectra.

present hosts, Table 2) and the barely studied blue emitter Fir6 (HOMO = -5.66 eV/LUMO = -2.32 eV, E_T = 2.70 eV dispersed in the present hosts, Table 2). Usually, when both the HOMO/LUMO gap and E_T of the phosphor increase, the PhOLED efficiency dramatically drops. In a multi-layer PhOLED, this can be compensated by the transporting and blocking layers but in single-layer device, this is far more difficult. To the best of our knowledge, only a few examples of blue SL-PhOLEDs (using a single host) is reported and all of them use the sky blue emitter Firpic.^{24,35–37,83} Indeed, the other blue dopants found

in OLEDs literature, for example FCNirpic (HOMO = -5.87 eV/LUMO = -2.65 eV, see ESI†)^{84,85} and Fir6^{40,53,54} (used in this study) are exclusively found as emitter in multi-layer PhOLEDs and not in SL-PhOLEDs using a single host (note that one example is reported but using a different host/co-host strategy⁴⁷).

Thus, with Firpic as sky blue emitter, **SPA-3,6-F(POPh)₂** displays this time the lowest performance with a low EQE of 6.5% and a high V_{on} of 3.5 V in accordance with a bad charges recombination and injection. The EQE is interestingly increased to 8.6% with **SPA-2-FPOPh₂** and the V_{on} is decreased

to 2.8 V signing a better charges injection, transport and recombination than in **SPA-3,6-F(POPh₂)₂**. It is nevertheless difficult to assign why **SPA-2-FPOPh₂** displays a higher performance than **SPA-3,6-F(POPh₂)₂** as the opposite was observed for the two other phosphors presented above. The case of **SPA-2,7-F(POPh₂)₂** is very different as it displays with FIrpic excellent performances. Indeed, a very high EQE of 18.0% ($CE = 39.0 \text{ cd A}^{-1}$ and $PE = 38.4 \text{ lm W}^{-1}$) was measured at 0.04 mA cm^{-2} (Table 3) and a maximum luminance of 8030 cd m^{-2} at 80 mA cm^{-2} was reached. We assign the very high efficiency of **SPA-2,7-F(POPh₂)₂** to the combination of many parameters. The high and well balanced mobilities of charge carriers are surely involved in this high performance as it is a key point when designing a host material for SL-PhOLED. It is also important to state that the LUMO levels of FIrpic and **SPA-2,7-F(POPh₂)₂** (Fig. 6 and Tables 2, 3) are very close and can also be involved in the present high performance. For these sky blue SL-PhOLEDs, the trend in term of V_{on} is similar to those exposed above for the other phosphors. Note that the V_{on} of **SPA-2,7-F(POPh₂)₂** based device is very low, 2.5 V, signing an efficient charges injection in the EML and well-balanced hole and electron mobilities. In a similar device configuration and as far as we are aware, only one host previously reported in literature displays a higher performance with FIrpic as emitter (EQE of 20.3%).³⁶ This molecule, reported by Liu and coworkers, incorporates an electron poor bis(diphenylphosphine oxide)-fluorene unit and a pendant diphenyl amine,⁸⁶ possessing therefore a similar molecular structure than that described herein. The higher performance obtained with this host compared to that of **SPA-2,7-F(POPh₂)₂** can be assigned to a better hole and electron mobilities balance. One can nevertheless note that our FIrpic-based devices display lower V_{on} (2.5 vs. 3 V), translating the electrochemical gap difference between the two molecules (2.79 vs. 2.86 eV).

Finally, in order to go deeper in the versatility of the present hosts and to reach devices emitting at shorter wavelength, blue phosphor FIr6 (HOMO = -5.66 eV /LUMO = -2.32 eV , $E_T = 2.70 \text{ eV}$ dispersed in the present hosts) was successfully used as emitter in identical SL-PhOLEDs. We were first surprised to note that **SPA-2-FPOPh₂** displays this time the highest efficiency in the series with an interesting maximal EQE of 9.1%. This value is even higher than that recorded with FIrpic. Note that the LUMO level of **SPA-2-FPOPh₂** is very close to that of FIr6 (-2.23 vs. -2.32 eV) and can be the explanation for this performance (the same observation, *i.e.* very close LUMO levels between host and guest, was also done with **SPA-2,7-F(POPh₂)₂** and FIrpic above). Surprisingly, the EQE of **SPA-2,7-F(POPh₂)₂** is very low, 6.5%, whereas this host was the most efficient for all the other guests. We can tentatively assigned this feature to its relatively low E_T , 2.76 eV, which is very close to that of FIr6 and back energy transfer may occur decreasing the PhOLED performance. Finally, as observed for FIrpic, the efficiency of **SPA-3,6-F(POPh₂)₂** is very low. The EQE reaches indeed 6.5% at a low current density but strongly decreases as the current density increases, showing the very bad performance of this host with this guest emitter. Compared to the only example of

SL-PhOLED using FIr6 found in literature (which is nevertheless different as a host/co-host is used in the EML),⁴⁷ **SPA-2-FPOPh₂** displays significantly improved performance (9.1 vs. 6.5%), clearly highlighting the efficiency of the chemical design strategy used herein. To conclude, these data show the real difficulty to design versatile hosts, which can be efficiently used with RGB phosphors. As the thickness of the EML has not been optimized for this last phosphor, it is obvious that the encouraging performance observed with **SPA-2-FPOPh₂** will be easily overpassed in the future.

It should be finally mentioned that all the devices exhibited identical red, green or blue emission arising exclusively from their corresponding iridium complex (and are therefore independent of the matrix used), showing an efficient energy transfer cascade (see electroluminescent spectra for all the device in Fig. 7). This is confirmed by the study of the emission spectra of the EML (host + guest 10% without electrodes), which are identical to the corresponding electroluminescent spectra (see ESI†). The CIE chromatic coordinates of the electroluminescent spectra are reported in Table 3. It is particularly important to note that FIrpic based devices display CIE coordinates of (0.15, 0.37) whereas those of FIr6 are shifted to (0.15, 0.30) confirming a more blue emission for the latter.

Conclusion

In this work, we have investigated different phenylacridine/diphenylphosphineoxide associations within a single host for RGB SL-PhOLEDs. The synthetic strategy developed is short, versatile, efficient, and uses cheap starting materials. For the future of the OLED technology, reducing the synthetic chemistry steps is highly required to reduce its environmental footprint. This work shows how the number and the position of the diphenylphosphine oxide units attached to the spiro-phenylacridine-fluorene backbone significantly impact the physical and electronic properties.

The three molecules investigated herein have been incorporated as host material in simplified single-layer PhOLEDs with four different iridium complexes emitting in the red, green and blue regions. With these conditions, we have shown how each parameter of the host (HOMO/LUMO energy levels, E_T , charge carriers mobility) influences the emission efficiency of the guest phosphor within the device and how a subtle combination of these three parameters is required. In addition, this work also reports the first examples of SL-PhOLEDs using the blue emitter FIr6. With this phosphor, the mono-substituted **SPA-2-FPOPh₂** displays the best performance in the series with a maximal EQE of 9.1% and a low V_{on} below 3 V. These first results appear very promising and deserve device optimizations in the future. As simplifying the device structure can be a central feature in the future of OLEDs, designing efficient and stable semi-conductors for this purpose is an important step. However, more researches are undoubtedly needed not only in term of molecular design but also in term of device engineering. Recently, it has been shown that specific treatment of ITO

(using chlorinated ITO followed by UV ozone treatment as anode) is an efficient technique to improve the performance of SL-PHOLEDs.³² Combining the best device architectures with the best host materials may result to very high efficiency SL-PhOLEDs.

Conflicts of interest

There are no conflicts to declare.

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References

- M. A. Baldo, D. F. O'Brien, Y. You, A. Shoustikov, S. Sibley, M. E. Thompson and S. R. Forrest, *Nature*, 1998, **395**, 151–154.
- Y. Tao, C. Yang and J. Qin, *Chem. Soc. Rev.*, 2011, **40**, 2943–2970.
- K. S. Yook and J. Y. Lee, *Adv. Mater.*, 2014, **26**, 4218–4233.
- C. Poriol and J. Rault-Berthelot, *J. Mater. Chem. C*, 2017, **5**, 3869–3897.
- Q. Wang, F. Lucas, C. Quinton, Y.-K. Qu, J. Rault-Berthelot, O. Jeannin, S.-Y. Yang, F.-C. Kong, S. Kumar, L.-S. Liao, C. Poriol and Z.-Q. Jiang, *Chem. Sci.*, 2020, **11**, 4887–4894.
- A. Maheshwaran, V. G. Sree, H.-Y. Park, H. Kim, S. H. Han, J. Y. Lee and S.-H. Jin, *Adv. Funct. Mater.*, 2018, **28**, 1802945.
- W. Li, J. Li, D. Liu and Q. Jin, *ACS Appl. Mater. Interfaces*, 2016, **8**, 22382–22391.
- J.-J. Huang, Y.-H. Hung, P.-L. Ting, Y.-N. Tsai, H.-J. Gao, T.-L. Chiu, J.-H. Lee, C.-L. Chen, P.-T. Chou and M.-K. Leung, *Org. Lett.*, 2016, **18**, 672–675.
- L. Ding, S.-C. Dong, Z.-Q. Jiang, H. Chen and L. S. Liao, *Adv. Funct. Mater.*, 2015, **25**, 645–650.
- L.-S. Cui, Y.-M. Xie, Y.-K. Wang, C. Zhong, Y.-L. Deng, X.-Y. Liu, Z.-Q. Jiang and L.-S. Liao, *Adv. Mater.*, 2015, **27**, 4213–4217.
- K. Udagawa, H. Sasabe, C. Cai and J. Kido, *Adv. Mater.*, 2014, **26**, 5062–5066.
- Y. Im, S. Y. Byun, J. H. Kim, D. R. Lee, C. S. Oh, K. S. Yook and J. Y. Lee, *Adv. Funct. Mater.*, 2017, **27**, 1603007.
- C. W. Lee and J. Y. Lee, *Adv. Mater.*, 2013, **25**, 5450–5454.
- X. Tang, X.-Y. Liu, Y. Yuan, Y.-J. Wang, H.-C. Li, Z.-Q. Jiang and L.-S. Liao, *ACS Appl. Mater. Interfaces*, 2018, **10**, 29840–29847.
- M. Kim and J. Y. Lee, *ACS Appl. Mater. Interfaces*, 2014, **6**, 14874–14880.
- X.-Y. Liu, X. Tang, Y. Zhao, D. Zhao, J. Fan and L.-S. Liao, *ACS Appl. Mater. Interfaces*, 2018, **10**, 1925–1932.
- K.-H. Kim, S. Lee, C.-K. Moon, S.-Y. Kim, Y.-S. Park, J.-H. Lee, J. Woo Lee, J. Huh, Y. You and J.-J. Kim, *Nat. Commun.*, 2014, **5**, 4769.
- W.-C. Chen, Y. Yuan, Z.-L. Zhu, Z.-Q. Jiang, S.-J. Su, L.-S. Liao and C.-S. Lee, *Chem. Sci.*, 2018, **9**, 4062–4070.
- C. Poriol and J. Rault-Berthelot, *Adv. Funct. Mater.*, 2020, **30**, 1910040.
- W.-Y. Hung, T.-C. Tsai, S.-Y. Ku, L.-C. Chi and K.-T. Wong, *Phys. Chem. Chem. Phys.*, 2008, **10**, 5822–5825.
- X. Qiao, Y. Tao, Q. Wang, D. Ma, C. Yang, L. Wang, J. Qin and F. Wang, *J. Appl. Phys.*, 2010, **108**, 034508.
- J. Ye, Z. Chen, K. Wang, F. An, Y. Yuan, W. Chen, Q. Yang, X. Zhang and C.-S. Lee, *Chem. – Eur. J.*, 2014, **20**, 13762–13769.
- C. Zang, X. Peng, H. Wang, Z. Yu, L. Zhang, W. Xie and H. Zhao, *Org. Electron.*, 2017, **50**, 106–114.
- Z. Liu, M. G. Helander, Z. Wang and Z. Lu, *Org. Electron.*, 2009, **10**, 1146–1151.
- Q. Jiang, Y. Xu, T. Yu, X. Qiu, R. Zhao, D. Zhao, N. Zheng, D. Hu, Z. Xie and Y. Ma, *New J. Chem.*, 2019, **43**, 6721–6727.
- C.-H. Chen, W.-S. Huang, M.-Y. Lai, W.-C. Tsao, J. T. Lin, Y.-H. Wu, T.-H. Ke, L.-Y. Chen and C.-C. Wu, *Adv. Funct. Mater.*, 2009, **19**, 2661–2670.
- M.-Y. Lai, C.-H. Chen, W.-S. Huang, J. T. Lin, T.-H. Ke, L.-Y. Chen, M.-H. Tsai and C.-C. Wu, *Angew. Chem., Int. Ed.*, 2008, **47**, 581–585.
- J. P. J. Markham, S.-C. Lo, S. W. Magennis, P. L. Burn and I. D. W. Samuel, *Appl. Phys. Lett.*, 2002, **80**, 2645–2647.
- W.-Y. Hung, T.-C. Wang, H.-C. Chiu, H.-F. Chen and K.-T. Wong, *Phys. Chem. Chem. Phys.*, 2010, **12**, 10685–10687.
- B. Huang, W. Jiang, J. Tang, X. Ban, R. Zhu, H. Xu, W. Yang and Y. Sun, *Dyes Pigm.*, 2014, **101**, 9–14.
- S. Thiery, D. Tondelier, B. Geffroy, E. Jacques, M. Robin, R. Métivier, O. Jeannin, J. Rault-Berthelot and C. Poriol, *Org. Lett.*, 2015, **17**, 4682–4685.
- Z. Wu, Z. Yang, K. Xue, C. Fei, F. Wang, M. Yan, H. Zhang, D. Ma and W. Huang, *RSC Adv.*, 2018, **8**, 11255–11261.
- F. Lucas, O. A. Ibraikulov, C. Quinton, L. Sicard, T. Heiser, D. Tondelier, B. Geffroy, N. Leclerc, J. Rault-Berthelot and C. Poriol, *Adv. Opt. Mater.*, 2020, **8**, 1901225.
- A. Yoshii, Y. Onaka, K. Ikemoto, T. Izumi, S. Sato, H. Kita, H. Taka and H. Isobe, *Chem. – Asian J.*, 2020, **15**, 2181–2186.
- H.-H. Chang, W.-S. Tsai, C.-P. Chang, N.-P. Chen, K.-T. Wong, W.-Y. Hung and S.-W. Chen, *Org. Electron.*, 2011, **12**, 2025–2032.
- F.-M. Hsu, L.-J. Chien, K.-T. Chen, Y.-Z. Li and S.-W. Liu, *Org. Electron.*, 2014, **15**, 3327–3332.
- Y. Yin, X. Wen, J. Yu, L. Zhang and W. Xie, *IEEE Photonics Technol. Lett.*, 2013, **25**, 1041–1135.
- Y. Yin, X. Piao, Y. Wang, J. Liu, K. Xu and W. Xie, *Appl. Phys. Lett.*, 2012, **101**, 063306.
- K. S. Yook and J. Y. Lee, *Adv. Mater.*, 2012, **24**, 3169–3190.

- 40 K. Gao, K. Liu, X.-L. Li, X. Cai, D. Chen, Z. Xu, Z. He, B. Li, Z. Qiao, D. Chen, Y. Cao and S.-J. Su, *J. Mater. Chem. C*, 2017, **5**, 10406–10416.
- 41 C.-C. Lai, M.-J. Huang, H.-H. Chou, C.-Y. Liao, P. Rajamalli and C.-H. Cheng, *Adv. Funct. Mater.*, 2015, **25**, 5548–5556.
- 42 X.-D. Zhu, Y.-L. Zhang, Y. Yuan, Q. Zheng, Y.-J. Yu, Y. Li, Z.-Q. Jiang and L.-S. Liao, *J. Mater. Chem. C*, 2019, **7**, 6714–6720.
- 43 Y. Zhao, C. Wu, P. Qiu, X. Li, Q. Wang, J. Chen and D. Ma, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2635–2643.
- 44 W. Song, L. Shi, L. Gao, P. Hu, H. Mu, Z. Xia, J. Huang and J. Su, *ACS Appl. Mater. Interfaces*, 2018, **10**, 5714–5722.
- 45 Y. Miao, K. Wang, L. Gao, B. Zhao, H. Wang, F. Zhu, B. Xu and D. Ma, *J. Mater. Chem. C*, 2018, **6**, 8122–8134.
- 46 W. Jiang, L. Duan, D. Zhang, G. Dong, L. Wang and Y. Qiu, *J. Mater. Chem.*, 2010, **20**, 6131–6137.
- 47 C. Fan, Y. Li, C. Yang, H. Wu, J. Qin and Y. Cao, *Chem. Mater.*, 2012, **24**, 4581–4587.
- 48 K. H. Yeoh, N. A. Talik, T. J. Whitcher, C. Y. B. Ng and K. L. Woon, *J. Phys. D: Appl. Phys.*, 2014, **47**, 205103.
- 49 S. E. Jang and J. Y. Lee, *J. Lumin.*, 2011, **131**, 2788–2791.
- 50 N. C. Erickson and R. J. Holmes, *Appl. Phys. Lett.*, 2010, **97**, 083308.
- 51 J. Y. Xue, T. Izumi, A. Yoshii, K. Ikemoto, T. Koretsune, R. Akashi, R. Arita, H. Taka, H. Kita, S. Sato and H. Isobe, *Chem. Sci.*, 2016, **7**, 896–904.
- 52 K. Ikemoto, A. Yoshii, T. Izumi, H. Taka, H. Kita, J. Y. Xue, R. Kobayashi, S. Sato and H. Isobe, *J. Org. Chem.*, 2016, **81**, 662–666.
- 53 C. Wu, B. Wang, Y. Wang, J. Hu, J. Jiang, D. Ma and Q. Wang, *J. Mater. Chem. C*, 2019, **7**, 558–566.
- 54 H.-H. Chou and C.-H. Cheng, *Adv. Mater.*, 2010, **22**, 2468–2471.
- 55 C. Poriol, L. Sicard and J. Rault-Berthelot, *Chem. Commun.*, 2019, **55**, 14238–14254.
- 56 C. Poriol and J. Rault-Berthelot, *Acc. Chem. Res.*, 2018, **51**, 1818–1830.
- 57 J.-D. Peltier, B. Heinrich, B. Donnio, E. Jacques, J. Rault-Berthelot and C. Poriol, *ACS Appl. Mater. Interfaces*, 2017, **9**, 8219–8232.
- 58 J.-D. Peltier, B. Heinrich, B. Donnio, O. Jeannin, J. Rault-Berthelot and C. Poriol, *Chem. – Eur. J.*, 2017, **23**, 17290–17303.
- 59 S. Bebiche, P. Cisneros-Perez, T. Mohammed-Brahim, M. Harnois, J. Rault-Berthelot, C. Poriol and E. Jacques, *Mater. Chem. Front.*, 2018, **2**, 1631–1641.
- 60 L. Sicard, C. Quinton, J.-D. Peltier, D. Tondelier, B. Geffroy, U. Biapo, R. Métivier, O. Jeannin, J. Rault-Berthelot and C. Poriol, *Chem. – Eur. J.*, 2017, **23**, 7719–7723.
- 61 S. Thiery, C. Declairieux, D. Tondelier, G. Seo, B. Geffroy, O. Jeannin, R. Métivier, J. Rault-Berthelot and C. Poriol, *Tetrahedron*, 2014, **70**, 6337–6351.
- 62 C. Quinton, L. Sicard, O. Jeannin, N. Vanthuyne and C. Poriol, *Adv. Funct. Mater.*, 2018, **28**, 180340.
- 63 M. Romain, D. Tondelier, B. Geffroy, O. Jeannin, E. Jacques, J. Rault-Berthelot and C. Poriol, *Chem. – Eur. J.*, 2015, **21**, 9426–9439.
- 64 I. Bulut, P. Chavez, S. Fall, S. Mery, B. Heinrich, J. Rault-Berthelot, C. Poriol, P. Leveque and N. Leclerc, *RSC Adv.*, 2016, **6**, 25952–25959.
- 65 M. Romain, D. Tondelier, O. Jeannin, B. Geffroy, J. Rault-Berthelot and C. Poriol, *J. Mater. Chem. C*, 2015, **3**, 97010–97014.
- 66 M. Romain, M. Chevrier, S. Bebiche, T. Mohammed-Brahim, J. Rault-Berthelot, E. Jacques and C. Poriol, *J. Mater. Chem. C*, 2015, **3**, 5742–5753.
- 67 M. Romain, D. Tondelier, B. Geffroy, A. Shirinskaya, O. Jeannin, J. Rault-Berthelot and C. Poriol, *Chem. Commun.*, 2015, **51**, 1313–1315.
- 68 L. J. Sicard, H.-C. Li, Q. Wang, X.-Y. Liu, O. Jeannin, J. Rault-Berthelot, L.-S. Liao, Z.-Q. Jiang and C. Poriol, *Angew. Chem., Int. Ed.*, 2019, **58**, 3848–3853.
- 69 C. Quinton, S. Thiery, O. Jeannin, D. Tondelier, B. Geffroy, E. Jacques, J. Rault-Berthelot and C. Poriol, *ACS Appl. Mater. Interfaces*, 2017, **9**, 6194–6206.
- 70 C. Poriol, J. Rault-Berthelot, S. Thiery, C. Quinton, O. Jeannin, U. Biapo, B. Geffroy and D. Tondelier, *Chem. – Eur. J.*, 2016, **22**, 17930–17935.
- 71 S. Thiery, D. Tondelier, C. Declairieux, B. Geffroy, O. Jeannin, R. Métivier, J. Rault-Berthelot and C. Poriol, *J. Phys. Chem. C*, 2015, **119**, 5790–5805.
- 72 D. H. Ahn, S. W. Kim, H. Lee, I. J. Ko, D. Karthik, J. Y. Lee and J. H. Kwon, *Nat. Photonics*, 2019, **13**, 540–546.
- 73 E. Baranoff and B. F. E. Curchod, *Dalton Trans.*, 2015, **44**, 8318–8329.
- 74 W.-H. Choi, G. Tan, W.-Y. Sit, C.-L. Ho, C. Y.-H. Chan, W. Xu, W.-Y. Wong and S.-K. So, *Org. Electron.*, 2015, **24**, 7–11.
- 75 J.-H. Seo, N.-S. Han, H.-S. Shim, J.-H. Kwon and J.-K. Song, *Bull. Korean Chem. Soc.*, 2011, **32**, 1415–1418.
- 76 G. Tan, S. Chen, N. Sun, Y. Li, D. Fortin, W.-Y. Wong, H.-S. Kwok, D. Ma, H. Wu, L. Wang and P. D. Harvey, *J. Mater. Chem. C*, 2013, **1**, 808–821.
- 77 R. J. Holmes, B. W. D'Andrade, S. R. Forrest, X. Ren, J. Li and M. E. Thompson, *Appl. Phys. Lett.*, 2003, **83**, 3818–3820.
- 78 Y. L. Chang, Z. B. Wang, M. G. Helander, J. Qiu, D. P. Puzzo and Z. H. Lu, *Org. Electron.*, 2012, **13**, 925–931.
- 79 J. P. Duan, P. P. Sun and C. H. Cheng, *Adv. Mater.*, 2003, **15**, 224–228.
- 80 R. Mertens, *The OLED Handbook: A Guide to OLED Technology, Industry & Market*, 2019th edn, 2019.
- 81 K. Klimes, Z.-Q. Zhu and J. Li, *Adv. Funct. Mater.*, 2019, **29**, 1903068.
- 82 S. Hu, J. Zeng, X. Zhu, J. Guo, S. Chen, Z. Zhao and B. Z. Tang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 27134–27144.
- 83 Y. Liu, L.-S. Cui, M.-F. Xu, X.-B. Shi, D.-Y. Zhou, Z.-K. Wang, Z.-Q. Jiang and L. S. Liao, *J. Mater. Chem. C*, 2014, **2**, 2488–2495.
- 84 S. O. Jeon, S. E. Jang, H. S. Son and J. Y. Lee, *Adv. Mater.*, 2011, **23**, 1436–1441.
- 85 S. O. Jeon, K. S. Yook, C. W. Joo and J. Y. Lee, *Adv. Funct. Mater.*, 2009, **19**, 3644–3649.
- 86 F.-M. Hsu, C.-H. Chien, C.-F. Shu, C.-H. Lai, C.-C. Hsieh, K.-W. Wang and P.-T. Chou, *Adv. Funct. Mater.*, 2009, **19**, 2834–2843.