

PAPER

[View Article Online](#)
[View Journal](#) | [View Issue](#)Cite this: *J. Mater. Chem. A*, 2020, **8**, 13340

Nickel confined in 2D earth-abundant oxide layers for highly efficient and durable oxygen evolution catalysts†

Yayun Pu,^{ab} Matthew J. Lawrence,^a Veronica Celorrio,^c Qi Wang,^d Meng Gu,^d Zongzhao Sun,^b Leonardo Agudo Jácome,^e Andrea E. Russell,^f Limin Huang^{*,b} and Paramaconi Rodriguez^{*,a}

Low cost, high-efficiency catalysts towards water splitting are urgently required to fulfil the increasing demand for energy. In this work, low-loading (<20 wt%) Ni-confined in layered metal oxide anode catalysts (birnessite and lepidocrocite titanate) have been synthesized by facile ion exchange methodology and subjected to systematic electrochemical studies. It was found that Ni-intercalated on K-rich birnessite (Ni-KMO) presents an onset overpotential (η_{onset}) as low as 100 mV and overpotential at 10 mA cm⁻² (η_{10}) of 206 mV in pH = 14 electrolyte. By combining electrochemical methods and X-ray absorption and emission spectroscopies (XAS and XES), we demonstrate Ni sites are the active sites for OER catalysis and that the Mn³⁺ sites facilitate Ni intercalation during the ion-exchange process, but display no observable contribution towards OER activity. The effect of the pH and the nature of the supporting electrolyte on the electrochemical performance was also evaluated.

Received 14th April 2020
Accepted 20th June 2020

DOI: 10.1039/d0ta04031b

rsc.li/materials-a

1. Introduction

The availability of a carbon neutral sustainable energy supply has become remarkably urgent in recent decades. Electrochemical water splitting, using renewable energy sources, offers an attractive route to chemical energy storage. However, one of the current limitations for the implementation of this technology is related to the inefficiency of the oxygen evolution reaction (OER) catalyst.^{1–3} Metal oxide catalysts based on precious metals such as IrO₂ and RuO₂ have been commonly used. However, their high cost and low abundance is a significant hindrance. Therefore, in the last decade there has been an enormous effort from the scientific community towards the

development of inexpensive and earth-abundant OER catalysts.^{4–9}

Amongst a number of different catalysts reported, layered metal oxide/hydroxide nanoparticles of earth abundant elements are of particular interest in electrochemical water splitting. Compared with the more typically employed RuO₂ and IrO₂ electrocatalysts, the M-based oxides/hydroxides (M = Ni, Co, Fe, Mn), as well as their oxyhydroxide and double metal hydroxides, are attractive as they possess extremely high surface areas, providing a large number of exposed active sites, and enhanced electronic conductivity after exfoliation.^{10–14} Such layered oxide/hydroxides consist of MO₆ host layers stabilized by charge-balancing ions accommodated into the interlayer spacing. The M–O layers fulfill both roles of skeleton, for architecture, and active sites, leading to insufficient atomic efficiency. To enhance the performance and efficiency of these catalysts previous reports have presented the preparation of ultrathin two-dimensional (2D) nanosheets, aiming at the exposure of more active sites.^{5,8} Improving the atomic efficiency of active atoms, such as Ru, Ir, Ni or Co, is economically attractive for large-scale industrial applications.^{15,16} In the literature a variety of strategies have been implemented, including synthesis of nanoclusters and even single atoms anchored on carbon-based and metal oxides-based supports.^{17,18} Although, to date, very little attention has been dedicated to the ions in the channels formed between the 2D layered oxide structures. Catalytic ions in between the layers can be arranged in-plane and immobilized by the opposite charge of the 2D host layer (Fig. 1a).¹⁹ The 2D channels fundamentally act

^aSchool of Chemistry, University of Birmingham, Edgbaston, Birmingham B15 2TT, UK. E-mail: P.B.Rodriguez@bham.ac.uk

^bDepartment of Chemistry, Southern University of Science and Technology, No. 1088 Xueyuan Blvd, Shenzhen, Guangdong 518055, China. E-mail: huanglm@sustech.edu.cn

^cDiamond Light Source Ltd, Diamond House, Harwell Campus, Didcot, OX11 0DE, UK

^dDepartment of Materials Science and Engineering, Southern University of Science and Technology, No. 1088 Xueyuan Blvd, Shenzhen, Guangdong 518055, China

^eBundesanstalt für Materialforschung und -prüfung (BAM), Department Materials Engineering, Unter Den Eichen 87, 12205 Berlin, Germany

^fSchool of Chemistry, University of Southampton, Highfield, Southampton, SO17 1BJ, UK

† Electronic supplementary information (ESI) available: Full description of the experimental methods, XRD patterns, calculation of charge integration, Tafel plots, long-term durability experiments and XAS fitting analysis. See DOI: 10.1039/d0ta04031b

as ideal electrochemical nanoreactors providing ample inter-layer spacing for free transportation of reactants and products and confinement of the under-coordinated active atoms.^{20,40} Previous work by Thenuwara *et al.*, show that Ni-intercalated on birnessite is a potential catalyst for the OER at pH = 14.⁴⁰

An important parameter affecting the electrochemical performance of catalysts during the OER is the pH. It is well known that the pH and composition of the electrolyte have a significant impact on the performance and durability of the catalyst during the OER. Establishing correlations between pH and measured voltammetric profiles facilitates the deduction of the active sites for OER and the evolution of adsorption/desorption processes and species on the surface. Takashima *et al.* reported that the instability of active Mn^{3+} at pH < 9 is the key factor in the observed poor activity of $\delta\text{-MnO}_2$ toward the

OER, which requires additional potential to electrochemically oxidize Mn^{2+} to Mn^{3+} .²¹ Koper's group proposed the surface deprotonation mechanism of NiOOH during OER catalysis based on the pH sensitivity observed.²² With that in mind, the catalytic activity of confined catalysts in layered structures, where reactant diffusion is rigorously restricted, would be strongly affected by the pH dominated deprotonation kinetics.

The manuscript describes the synthesis and full characterization of Ni ions confined to the inter-layer space in 2D layered MnO nanoparticles prepared *via* a facile ion exchange strategy using NiCl_2 as a precursor. The characterization of the materials using XAS/XES provided comprehensive understanding of the role of Mn^{3+} in the Ni intercalation process, the oxidation state of the species and the different Ni coordination/structures obtained for Ni-KMO and Ni-HMO. We demonstrate the catalytic

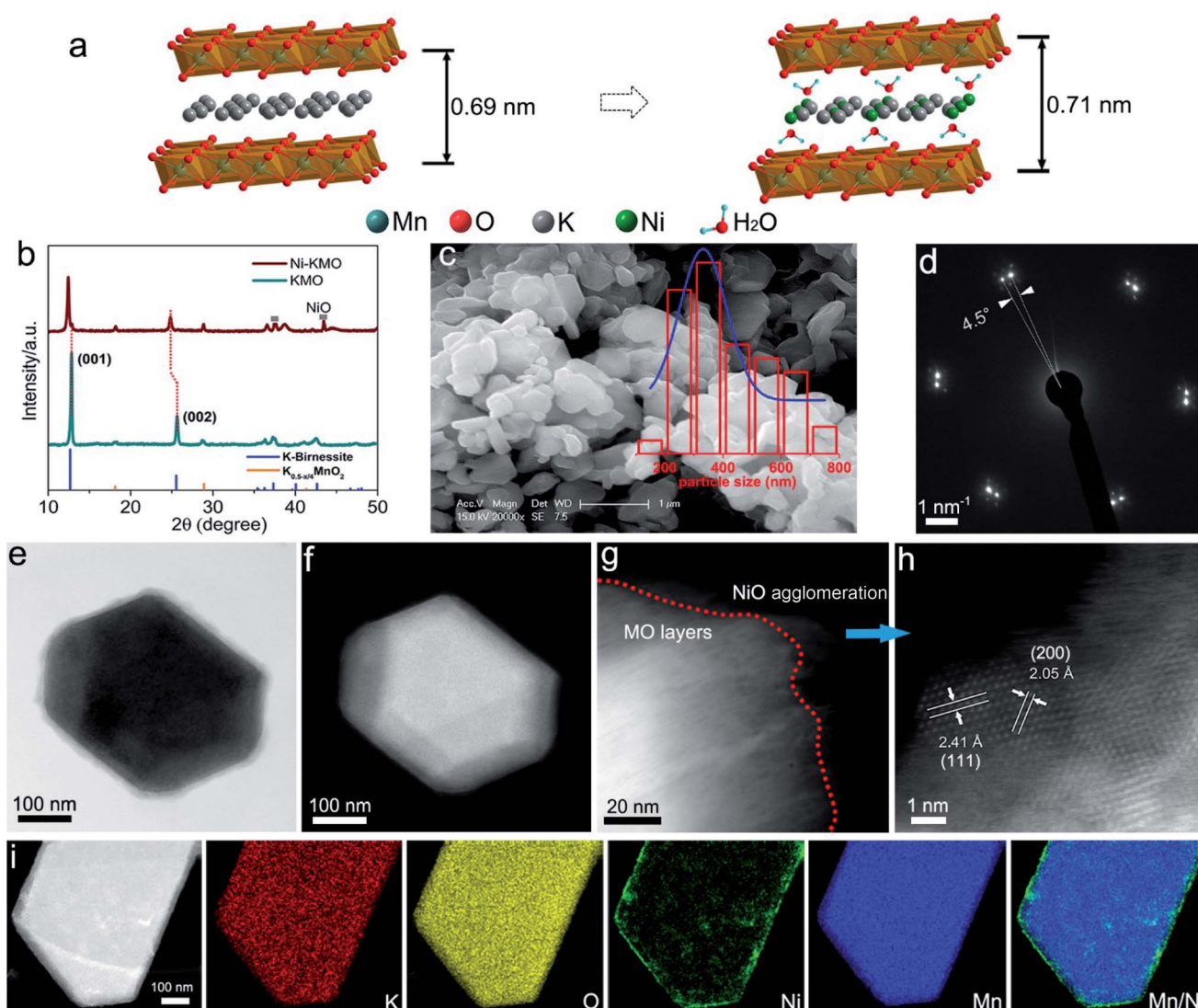


Fig. 1 (a) Schematic illustration of the impact of Ni intercalation into layered K-birnessite manganese oxide. (b) X-ray diffraction patterns of Ni-KMO and pristine KMO. (c) SE-SEM image of Ni-KMO nanoplates. The particle size distribution histogram is inset. (f) SAD diffraction pattern (ZA = $[001]_{\text{Ni-KMO}}$) of particle in (e). (e) Bright field (BF) and (f) HAADF-STEM images of Ni-KMO nanoparticle seen along $[001]$ zone axis (ZA). HAADF-STEM images (g), and (h) enlarged Ni agglomeration at the edge of a Ni KMO nanoplate. (i) HAADF-STEM image of single Ni-KMO nanoplate with corresponding EDS elemental mapping.



activity towards the OER and the long term durability of the catalyst. In addition, we also present the reaction mechanism for the OER on the Ni-confined catalyst based on an exhaustive study of the catalyst under different conditions, *e.g.* pH and scan rate.

2. Results and discussion

2.1 Characterization of Ni-confined KMO

Ni-exchanged $K_{0.45}MnO_2$ (Ni-KMO) was prepared by mixing K-rich birnessite ($K_{0.45}MnO_2$) and $NiCl_2$ solution at ambient temperature, as described in the Experimental section (ESI†). Fig. 1b presents the XRD patterns of the as-prepared KMO and after Ni exchange. From the XRD pattern it was determined that the K-birnessite possessed a layered structure with an interlayer spacing of 0.69 nm (AMCSD 0001302).²³ The same layered structure was retained after the Ni exchange process, with XRF measurements suggesting that Ni cations partially replaced K ions in the interlayer spacing (Table 1). Such intercalation resulted in the enlargement of the interlayer distances (001) and (002) along the *c* axis, demonstrated by the decrease in 2θ values for those diffraction signals at $2\theta = 12.43^\circ$ and 24.84° . The SEM (Fig. 1c) and STEM images (Fig. 1d, e and g–i) of Ni-KMO show plate-like nanoparticles with facets with dimensions between 300–400 nm with near-identical morphology to that of KMO (Fig. S1†). The bright field (BF) and high angle annular darkfield (HAADF) images in Fig. 1e and f show a single nanoparticle comprising stacked layers of KMO. In Fig. 1d, the selected area diffraction (SAD) pattern shows a splitting of the reflections, which can be interpreted as an in-plane rotation of 4.5° (marked at the Ni-KMO (110) reflection) between two sets of layers in the nanoparticle. The HAADF-STEM image in Fig. 1h suggests that the Mn–O framework retained a layered structure, which is consistent with the XRD results. In addition to the Ni atoms inserted into the interlayer channels, aggregation of Ni at the edges of the nanoplate crystals, generating a few nanometer-thick region, is observed in the HAADF images and EDS elemental maps (Fig. 1i (contrast the Ni and Mn HAADF images) and Fig. S2 and S3†). The additional XRD signals observed for Ni-KMO at 37.4° and 43.4° , which can be assigned to (111) and (200) of NiO (JCPDS 73-1519), are in agreement with STEM results.

2.2 XAS analysis

The adsorption energies of *OH and O^* are predominantly dependent on the electronic configuration of the Ni surfaces.²⁴ Therefore, the local environment of the Ni sites and associated species determine the corresponding electrochemical activity of

these materials. X-ray absorption spectroscopy (XAS) was implemented to determine the oxidation state and coordination environment of Ni and the Mn-oxide host structure and to enable correlation of these parameters with the underlying OER performance of Ni-KMO. The characteristic XANES spectra of metallic Mn, standard Mn-oxide compounds with oxidation states spanning the range Mn^{+2} to Mn^{+4} and the synthesized X–K(H)MO nanoplates (where X= H or Ni) are presented in Fig. 2a.

The edge positions, determined by the maximum of the first derivative, for Mn foil, MnO, Mn_3O_4 , Mn_2O_3 and MnO_2 were observed at 6539.0, 6544.2, 6546.7, 6547.7 and 6552.5 eV, respectively, in agreement with existing literature values.^{25–27} The measured edge positions for KMO, H(K)MO and Ni-KMO and Ni-HMO were 6549.3, 6550.3, 6549.6 and 6552.0 eV, respectively. The normalised absorption pre-edge features of the manganese oxide derivatives and relevant Mn K-edge reference oxide compounds are presented in Fig. S4.† An increase in pre-edge peak energy is observed with an increase in Mn oxidation state (Fig. S4a†). The nature of the pre-edge features of the spectra at the Mn K-edge suggest that there is a contribution from Mn^{2+} , Mn^{3+} and Mn^{4+} to the overall structure and the intermediate mean Mn oxidation state is between +3 and +4. Thus, X-ray emission spectroscopy (XES) was employed to obtain more accurate Mn oxidation state information. The measured XES spectra and corresponding fits, a function of the sum contributions of the standard Mn^{n+} oxide compounds, are presented in Fig. S5.† The mean Mn oxidation state of pristine KMO was determined as +3.9 (Table S1†). A decrease in mean oxidation state was observed upon cation exchange with both H^+ and Ni^{2+} , with the greatest decrease observed for H(K)MO. The fraction of Mn^{4+} is effectively the same for Ni-KMO and KMO, while the Mn^{3+} fraction decreases upon Ni-exchange. This suggests that Ni^{2+} intercalation is more favorable at those Mn^{3+} sites, which are commonly accompanied by defects that contribute to the overall negative charge of Mn–O layers.⁸ The defects with higher Gibbs energy favor the ion exchange reaction of K with Ni. The concentration of Mn^{3+} increases significantly in H(K)MO relative to KMO, thus more Ni cations were intercalated into Ni-HMO, as confirmed by the XRF analysis (Table 1).

The characteristic XANES spectra measured of metallic Ni, Ni-oxide reference compound, Ni-KMO, Ni-HTO and Ni-HMO are presented in Fig. 2b. The edge position of the Ni foil and NiO reference were determined as 8333.0 eV and 8345.2 eV, respectively. The absorption edges of Ni-KMO, Ni-HTO and Ni-HMO were determined as 8345.3, 8345.9 and 8346.0 eV respectively, suggesting a Ni oxidation state of +2. For Ni-KMO the XANES spectrum appears identical to that of the NiO reference, indicative of the presence of NiO structure in Ni-

Table 1 XRF results of Ni-KMO, Ni-H(K)MO and Ni-HTO^a

Samples	Ni (at%)	K (at%)	Mn (at%)	Ti (at%)
$Ni_{0.16}K_{0.2}MnO_2 \cdot 0.7H_2O$	11.67 ± 0.18	14.46 ± 0.2	73.87 ± 0.37	—
$Ni_{0.33}K_{0.04}MnO_2 \cdot 0.9H_2O$	24.1 ± 0.4	2.69 ± 0.09	71.26 ± 0.5	—
$Ni_{0.033}Ti_{0.91}O_2 \cdot 0.6H_2O$	3.60 ± 0.07	—	—	96.40 ± 0.07

^a TGA was performed to determine the water amount in Ni-intercalated layered metal oxides (Fig. S24).



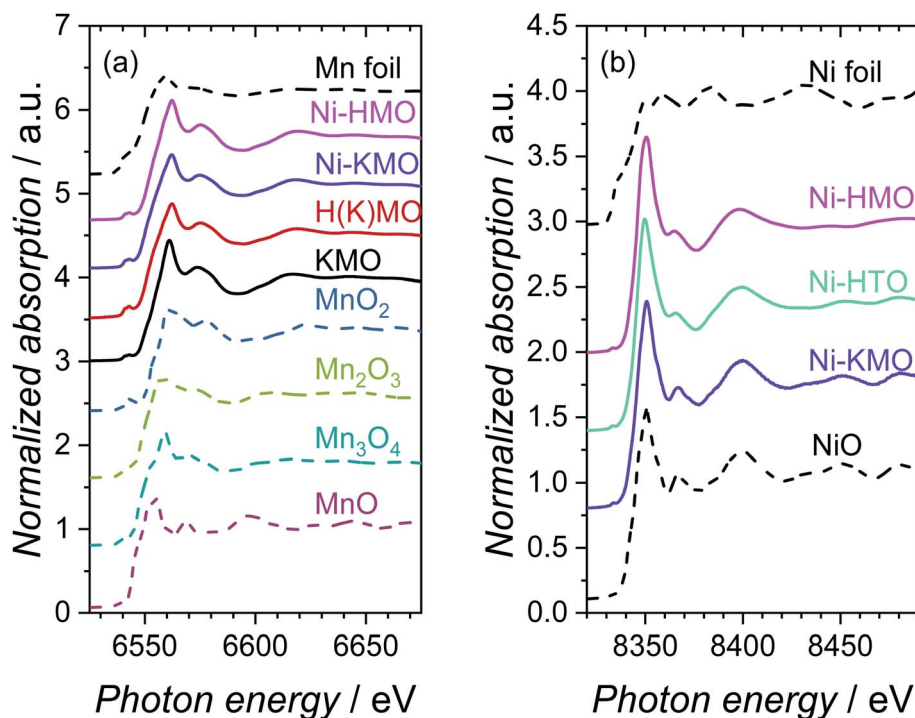


Fig. 2 Normalized XANES spectra measured at the (a) Mn K-edge and (b) Ni K-edge for KMO, H(K)MO, Ni-KMO, Ni-HMO and Ni-HTO catalysts, in addition to relevant standard oxide compounds and metal foils. References, represented by dashed lines, are shown for comparison.

KMO. While slightly different spectrum of Ni-HMO compared to Ni-KMO suggesting a different Ni^{2+} structure, as confirmed by the fitted EXAFS results below. There are no significant differences in pre-edge features observed at the Ni K-edge (Fig. S4b†).

The Mn K-edge EXAFS Fourier transforms (FTs) of the synthesised KMO, H(K)MO, Ni-KMO and Ni-HMO nanosheets and reference samples are presented in Fig. 3a. The first peak at $\sim 1.5 \text{ \AA}$ (not phase shift corrected) is due to the Mn–O scattering and the second peak at $\sim 2.5 \text{ \AA}$ (not phase shift corrected) is due

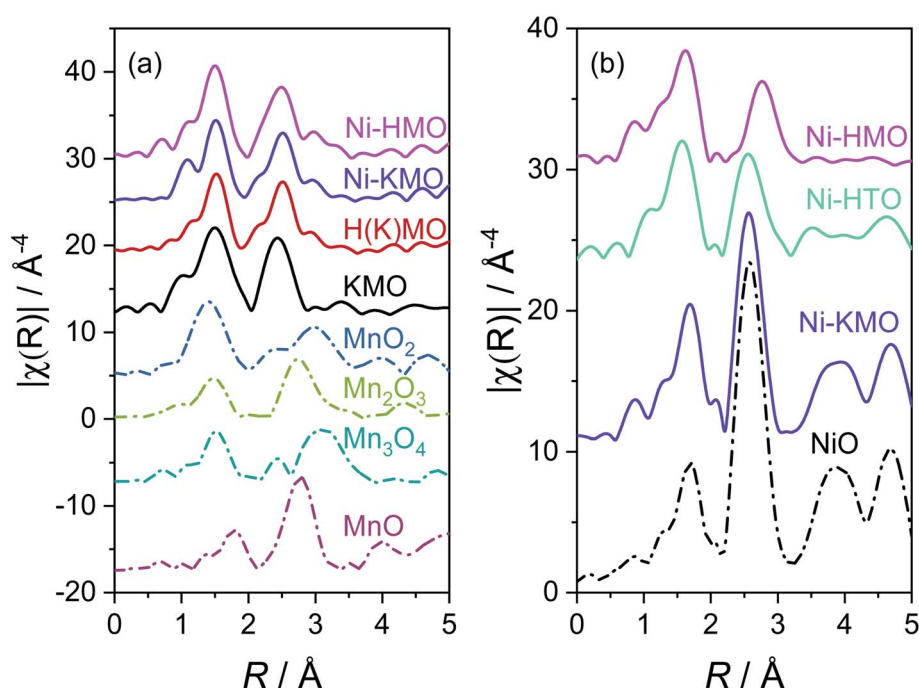


Fig. 3 EXAFS FTs of $k^3\chi(k)$ functions for KMO, H(K)MO, Ni-KMO, Ni-HMO and Ni-HMO nanosheets and standard oxide reference compounds at (a) Mn K-edge and (b) Ni K-edge. References are shown for comparison.



to Mn–Mn/Mn–Ni scattering (Tables S2, S3 and Fig. S6†). The two Mn scattering distances are generated by the two distinct Mn birnessite lattice sites: edges and face-centers. The Ni K-edge EXAFS FT of Ni-KMO is near-identical to the NiO reference compound. The first shell coordination at ~ 1.7 Å (not phase shift corrected) corresponds to Ni–O whilst the second shell coordination at ~ 2.6 Å (not phase shift corrected) corresponds to Ni–Ni scattering (Table S4†). The relative amplitude of the second shell peak for Ni-KMO compared to NiO suggests that the former comprises NiO nanoparticles. This is not the case for Ni-HTO and Ni-HMO, where the relative amplitude of the second shell peak differs from that of NiO, suggesting the formation of a different structure (Tables S5 and S7†).

In summary, the *ex situ* XAS/XES results clarify the importance of Mn³⁺ on the Ni-exchange intercalation. The Ni-content in each of the samples is proportional to the concentrations Mn³⁺ sites. For all the prepared samples, Ni is present in the oxidation state (+2), but different Ni coordination/structures were obtained for Ni-KMO and Ni-HMO.

3. Electrochemical performance of Ni-confined catalyst on KMO

3.1 Evaluation of the electrochemical activity and durability

The catalytic activity of Ni-KMO was evaluated over the pH range 8–14, to provide both assessment of the activity of the

electrocatalysts for the OER and enable deeper understanding of the mechanism behind this activity. Fig. 4a–c show the stable *iR*-corrected voltammetric profiles of the Ni-KMO catalyst as a function of the pH, with the voltammograms shown being those obtained after 30 consecutive cycles; the evolution of the curves over the 30 scans is presented in Fig. S10.† Current densities are reported as mA cm^{−2} normalized by the geometric area of the electrode. As can be seen, the OER onset overpotential (η_{onset}) increases with decreasing pH, from 130 mV at pH 14 to 220 mV at pH 12. (Note that all voltammetric responses are reported *vs.* RHE and, thus, differences are not attributable to changes to the potential of the reference electrode.). At the same time, the overpotential at 10 mA cm^{−2} (η_{10}) was 240 mV at pH 14, which is significantly smaller than the $\eta_{10} = 1090$ mV observed at pH 12. The marginal difference in measured η_{onset} and sharp contrast at η_{10} confirm the lower reaction rate of the OER at lower pH due to the lower concentration and availability of OH[−]. Fig. 4b summarizes the observed trends.

The redox activity of the confined Ni clusters is evident in the voltammograms as a quasi-reversible peak in the region between 1.2–1.5 V was observed at pH 14 and 12 (Fig. 4c). This peak has previously been ascribed to the Ni²⁺ ↔ Ni³⁺ couple of Ni based catalysts.^{15,22,28,29} Interestingly, at pH = 14 the oxidation peak of Ni²⁺ → Ni³⁺ appears at 1.24 V, which is significantly more cathodic relative to previous reports in the literature (~ 1.4 V *vs.* RHE on carbon electrode). This unusual cathodic

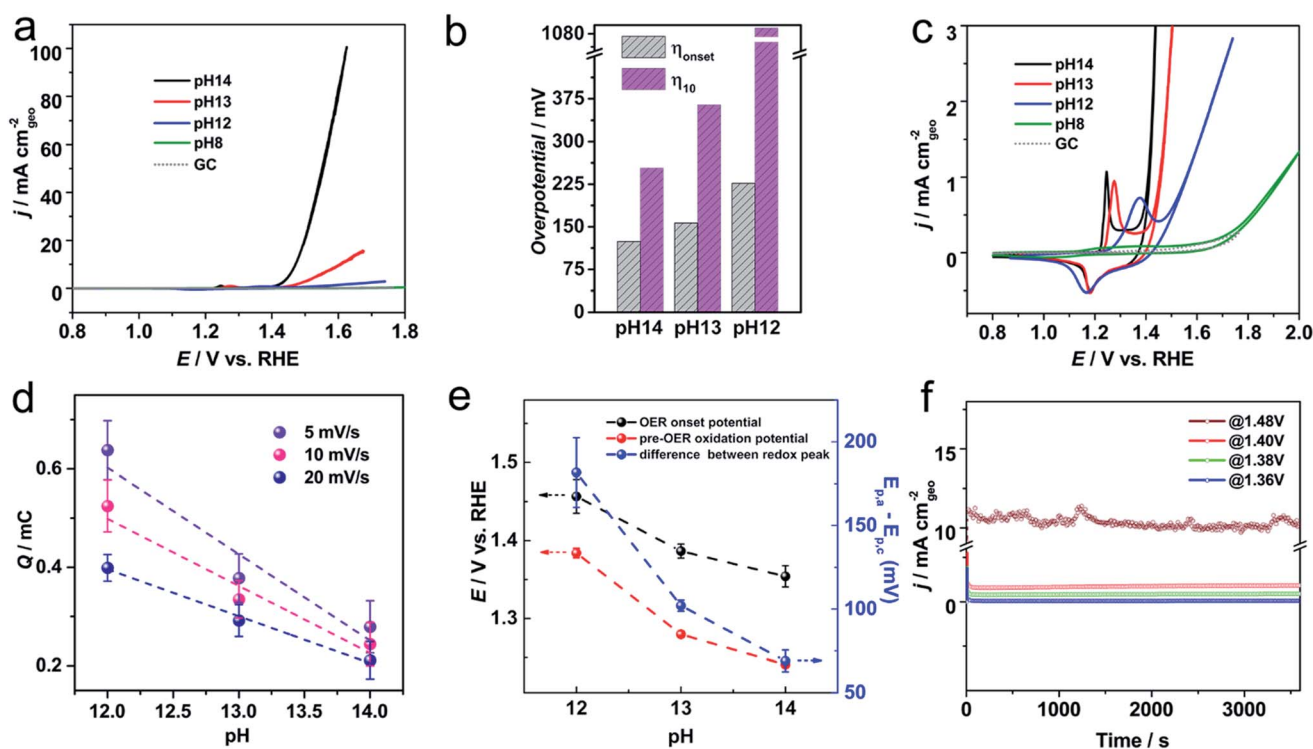


Fig. 4 (a) Stable cyclic voltammograms of Ni-KMO at various pHs after 30 consecutive scans in NaOH electrolyte, with scan rate of 20 mV s^{−1}. The profile of the glassy carbon (GC) support is included for comparison. All the curves are *iR*-corrected. (b) The onset overpotential (η_{onset}) and overpotential at $j = 10$ mA cm^{−2} (η_{10}) obtained from (a). (c) Enlarged region showing the Ni²⁺ → Ni³⁺ redox peaks at various pHs. (d) Integrated charge of Ni^{2+/3+} oxidation peak as a function of pH at various scan rates. (e) OER onset potential, pre-OER oxidation peak potential and Ni^{2+/3+} ΔE variation with pH. (f) Chronoamperometry profiles of Ni-KMO at pH = 14 at various applied potentials as indicated in the figure.



shift of oxidation peak to 1.24 V is beneficial for subsequent catalytic oxygen evolution. As seen in Fig. 4c, a direct relationship between the position of the $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ oxidation peak on the Ni-KMO and the OER onset potential was observed as a function of pH. Increasing the pH of the solution from 12 to 14 results in the shift of the $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ oxidation peak together with a shift of the onset potential of OER towards less positive potentials. This is a clear indicator that the Ni^{3+} are active sites for the OER. Moreover, the correlation between redox peaks and OER activity over multiple cycles is shown in Fig. S10.† At any one pH, the $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ oxidation peak shifts cathodically and the OER current density increases as a function of the number of cycles, confirming the role of $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ transformation on OER catalysis. This behavior is in agreement with previous work that established the relationship between the peak position of the redox couple on Ni-Fe(OOH) and the OER catalytic activity.³⁰

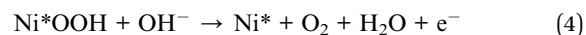
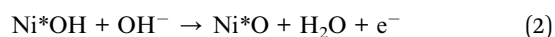
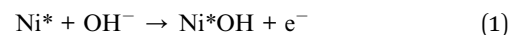
Since the OER catalytic activity is directly correlated to the nature of the Ni centers, understanding the redox peak of the Ni in the cyclic voltammograms may contribute to greater understanding of the reaction mechanism as well as in the development of better catalyst. Voltammetric profiles of the $\text{Ni}^{2+}/\text{Ni}^{3+}$ redox peaks were recorded at different pHs (Fig. S11†). At pH = 14 the presence of two discrete processes, Ox1/Re1 and Ox2/Re2, were observed. Previous reports assigned the process Ox1 \leftrightarrow Re1 to the phase transition $\alpha\text{-Ni}(\text{OH})_2/\gamma\text{-NiOOH}$, while the process labelled Ox2 \leftrightarrow Re2 has been associated with the transition of $\beta\text{-Ni}(\text{OH})_2$ to the active phase $\beta\text{-NiOOH}$.^{31,32} A schematic illustration of the phase transitions is presented in Fig. S12.† When the pH is decreased to 13, only one oxidation peak is observed (Ox1) and the peak labelled Re2 shifts towards more positive potentials and its current density decreases (Fig. S11†). At pH = 12, the Re2 peak is not observable and only the redox peak Ox1/Re1 is visible. The absence of the $\beta\text{-NiOOH}$ transformation at pH = 12 was accompanied by a significant reduction in measured OER activity.

To explore the underlying mechanism, we thoroughly examined the relationship between the $\text{Ni}^{2+}/\text{Ni}^{3+}$ redox peaks and the scan rates. The linear relationship between peak current density and scan rate at pH = 14 was indicative that the surface reaction kinetics were dominated by charge transfer steps, whereas the linear relationship between peak current density and the square root of the scan rate at pH = 12 was indicative that the diffusion limitation was the dominating factor (Fig. S13†).

The charge associated with the redox peaks as a function of the pH is also shown in Fig. 4d and S14.† As can be seen, opposite to the literature,³⁰ the charge associated to the redox peak of $\text{Ni}^{2+}/\text{Ni}^{3+}$ increases as the pH decreases. We further explore this phenomenon by analyzing the charges of the redox peak at pH = 12 and pH = 14 as a function of the number of cycles. Fig. S15† shows the comparison between the charge of the redox peaks of the 1st scan and 30th scan for the different pHs. The first observation is that slightly larger charges are observed at pH = 14 and pH = 13. As can be seen, the charge of the oxidation peak increases after electrochemical cycling across the pH range explored. However, the increase of the charge of the redox peak is significantly larger at pH = 12 and

after 30 cycles. Charge of the redox peak at pH = 12 (399 μC) is 1.89-fold that of the same peak at pH = 14 (211 μC). According to the discussion above, and the XAS results (Section 2.2) the Ni-clusters intercalated between the KMO layers must undergo an “activation process”, which is attributed to a phase transformation of the Ni oxide species. While at pH = 14, Ni sites with OH^- adsorption are fully transformed into the $\beta\text{-NiOOH}$ active form, at pH = 12 most of the catalyst is only transformed into the $\gamma\text{-NiOOH}$ phase. Considering that the oxidation state of the $\beta\text{-NiOOH}$ is +3 and the $\gamma\text{-NiOOH}$ is ~ 3.7 (Fig. S15†), the transformation from Ni^{2+} to $\gamma\text{-NiOOH}$ requires ~ 1.7 fold more charges than that to the $\beta\text{-NiOOH}$. Therefore, our results suggest that at pH = 14 the active phase for the OER is the $\beta\text{-NiOOH}$ and at pH = 12, the active phase for the OER is the $\gamma\text{-NiOOH}$. The unique structure of the Ni confined in 2D layer structure with flexibly adjustable interlayer spacing gives rise to such phenomenon.

In order to obtain more information on the reaction mechanism, Tafel analysis of the OER curves at different pHs was performed. Note that all voltammograms were *iR* corrected prior to the Tafel analysis. It has been previously proposed that the overall mechanism of OER catalysis on Ni sites in alkaline media involve four on-electron-transfer steps (eqn (1)–(4)).^{15,33} The Tafel plots for the OER on Ni-KMO at different pHs are presented in Fig. S16.† The Tafel slope, at pH = 14, is 60 mV dec^{-1} suggesting that the rate-determining step (RDS) was the deprotonation of adsorbed hydroxyl species (eqn (2)), in agreement with previous works on Ni hydroxide/oxide catalyst.³⁴ On the other hand, a Tafel slope of 232 mV dec^{-1} was obtained at pH 12, suggesting that the RDS at lower pH was the adsorption of OH^- species (eqn (1)). This can be rationalized by consideration of the nature of Ni sites. A higher concentration gradient is required to promote the transport of the OH^- species from electrolyte through the channels of the layered oxide structure to the confined Ni nanoreactor sites. This diffusion-dependent process gives rise to the significant difference in OER activity observed for Ni-KMO with varying pHs.



The evaluation of the electrochemical performance of a catalyst should not only be based on its catalytic activity but also on the durability of the catalyst. Long-term stability of the Ni-KMO was recorded at pH = 14 *via* chronoamperometry at various applied potentials (Fig. 2f). As can be seen in Fig. 2f, the current density during the OER remained stable over 1 h of electrolysis at different relevant potentials. In order to gain insight into the structural and compositional stability of the catalyst, SEM and HAADF-EDS images of the Ni-KMO catalyst were collected after 10 h (Fig. S17†). Interestingly, no significant changes in the morphology or the element distribution of the



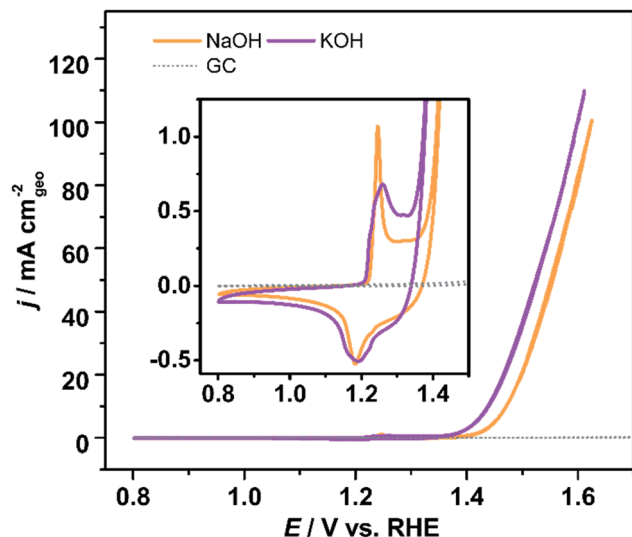


Fig. 5 Voltammetric profiles of Ni-KMO in 1 M KOH and NaOH electrolyte, as indicated in the figure, inset is the enlarged redox features. Glassy carbon shown for comparison. Scan rate $\nu = 20 \text{ mV s}^{-1}$.

catalyst were detected confirming the excellent stability of the catalyst after 10 h of electrolysis. A comparison of the long term stability, up to 23 h electrolysis in 1 M NaOH, of the Ni-KMO at 1.48 V and Ni-HTO at 1.51 V shows stable current densities for catalysis with current densities over 10 mA cm^{-2} (Fig. S23†).

3.2 Effect of the cations on the OER catalytic activity

Previous works have also demonstrated that in addition to the pH, the nature of the electrolyte cation also affects the observed activity and selectivity of electrochemical processes, especially in alkaline media.^{35,36} We investigated the effect of K^+ and Na^+ cations on the electrochemical activity of Ni-KMO towards the OER (Fig. 5). As can be seen, although a slightly different shape of the $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ redox peaks was observed, the charges for the $\text{Ni}^{2+}/\text{Ni}^{3+}$ redox peaks are almost identical, 0.261 mC in NaOH and 0.263 mC in KOH respectively. Ni-KMO presents lower η_{onset} and η_{10} for the OER in KOH ($\eta_{\text{onset}} = 100 \text{ mV}$ and $\eta_{10} = 206 \text{ mV}$) relative to the respective values in NaOH ($\eta_{\text{onset}} = 130 \text{ mV}$ and $\eta_{10} = 240 \text{ mV}$). It has been previously demonstrated, that the nature of the cations affects the catalytic activity of the OER and the ORR. In particular, those reports indicate that potassium enhances the catalytic activity.^{37–39} This effect has been associated to the non-covalent interaction between $\text{K}^+(\text{H}_2\text{O})_x/\text{Na}^+(\text{H}_2\text{O})_x$ clusters and the active Ni-oxygenated species at the surface Ni-OH*, Ni-O*, Ni-OOH*. Fig. 5 shows that the onset of the $\text{Ni}^{2+}/\text{Ni}^{3+}$ oxidation reaction is independent of the cation present in the electrolyte (KOH or NaOH) which indicates similar adsorption reaction of OH^- on Ni sites. On the other hand, at slightly more positive potentials a slight cathodic shift of OER onset in KOH suggests that the subsequent deprotonation step is favorable in KOH. Suntivich *et al.* suggested that charged $\text{K}^+(\text{H}_2\text{O})_x/\text{Na}^+(\text{H}_2\text{O})_x$ clusters would

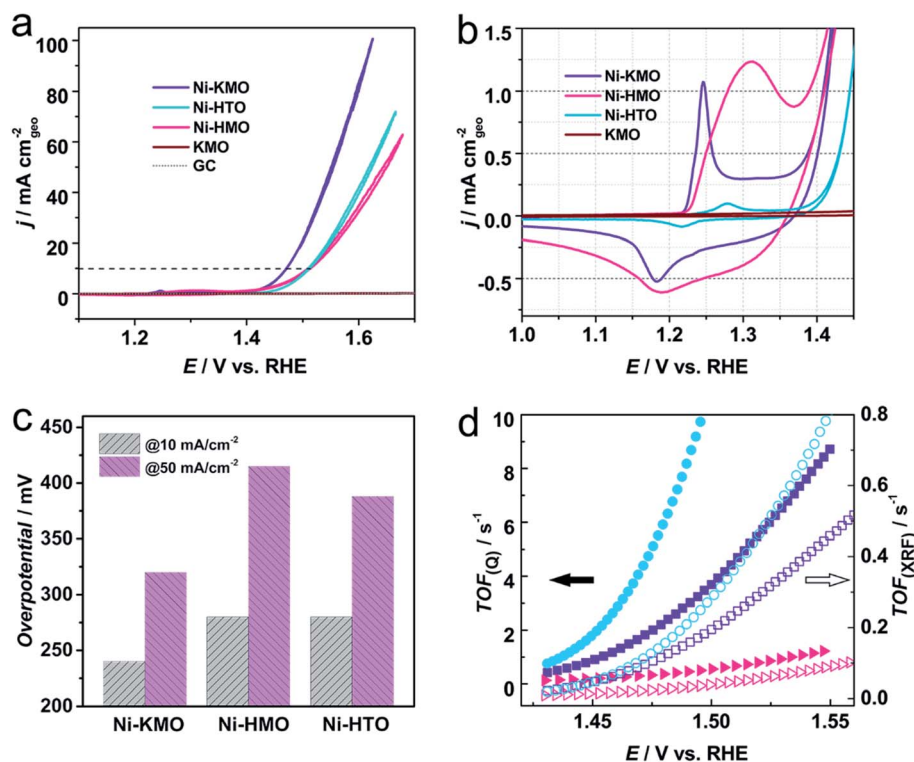


Fig. 6 (a) *iR*-corrected voltammetric profiles of Ni-KMO, Ni-HMO and Ni-HTO after 30 consecutive scans at pH = 14, in 1 M NaOH at 20 mV s^{-1} . The evolution of the voltammograms as function of the number of scans can be found in Fig. S22.† (b) Enlarged region showing the $\text{Ni}^{2+} \rightarrow \text{Ni}^{3+}$ redox peaks and OER onset. (c) η_{50} and η_{10} obtained from (a). (d) TOF as a function of potential calculated using the content of Ni content on Ni-KMO (square symbol), Ni-HMO (triangle symbol) and Ni-HTO (circle symbol) determined by the $\text{Ni}^{2+}/\text{Ni}^{3+}$ redox charge (solid symbols) and the XRF (empty symbols).



Table 2 Summary of electrochemical data of Ni-KMO, Ni-HMO, Ni-HTO

	$\text{Ni}_{0.16}\text{K}_{0.2}\text{MnO}_2 \cdot 0.7\text{H}_2\text{O}$	$\text{Ni}_{0.33}\text{K}_{0.04}\text{MnO}_2 \cdot 0.9\text{H}_2\text{O}$	$\text{Ni}_{0.033}\text{Ti}_{0.91}\text{O}_2 \cdot 0.6\text{H}_2\text{O}$
η_{10}/mV	240	280	280
Integration charge/mC	0.278	0.743	0.0342
Corresponding Ni content/ μg	0.169	0.452	0.020
Tafel slope/ mV dec^{-1}	60	113	50
Ni content [XRF]/wt%	8	16	2
j_{mss} at $\eta = 240 \text{ mV/A g}_{\text{Ni}}^{-1}$	602	140	710

modify the energy of M-oxygenated species, thus altering their stability.³⁷

3.3 Investigation of the coordination of active Ni on the catalytic activity

In order to further investigate the specific role of Ni coordination in the performance of the catalyst, Ni-exchanged HMnO_2 (Ni-HMO) and Ni-exchanged $\text{HTi}_{0.91}\text{O}_2$ (Ni-HTO) were prepared and evaluated towards the OER. Both materials, HMO and HTO, present the same layered structure (Fig. S18[†]), therefore any observable differences in the catalytic activity would suggest differences in the interaction between the Ni and species such as Mn/Ti and $\text{K}/\text{H}_2\text{O}$ around. Both Ni-HMO and Ni-HTO catalysts display notable OER activity (Fig. 6a) in comparison with the pristine HMO and HTO (Fig. S19[†]). As can be seen on Table S2,[†] Ni-KMO shows the greatest OER catalytic activity ($\eta_{10} = 240 \text{ mV}$) of the three Ni-exchanged layered oxide catalysts in regard to geometrical activity. Despite the remarkable difference of Ni redox peaks for the 3 catalysts, the OER η_{onset} values are nearly identical, indicating that confined Ni is truly the active component in the observed activity of Ni-exchanged HMO and HTO. Tafel slopes of three Ni confined samples are shown in Fig. S20.[†] Ni-HTO with 50 mV dec^{-1} is smaller than Ni-KMO (60 mV dec^{-1}) and Ni-HMO (113 mV dec^{-1}). These data suggest that the chemical deprotonation process is the RDS for both Ni-HTO and Ni-KMO, whilst for Ni-HMO it is the adsorption of OH^- . To obtain further information on the nature of the differences in the shape of the redox peaks and the catalytic activity, the Ni content in each of the catalysts was determined by XRF (Table 2). It was found that Ni-HMO had the greatest Ni

loading (16 wt%), followed by Ni-KMO (8 wt%) and Ni-HTO (2 wt%). Therefore, the differences in the shape of the redox peaks could be attributed to the differences in Ni loading and coordination, but the larger catalytic activity of the Ni-KMO could not be directly correlated to the Ni loading. This implies that Ni coordination plays the predominant role in water catalysis. This assumption is in line with the previous work by Thenuwara *et al.*, which suggested that highly ordered water molecules lower the potential barrier that facilitate the OER.⁴⁰

So far we have demonstrated that the Ni sites solely give rise to the observed OER activity of the Ni-confined layered metal oxide catalysts, thus, the Ni content allows us to determine the turnover frequency (TOF) of the catalyst, a key indicator of the intrinsic activity of a catalyst. However, two possible strategies can be adopted in this case. The TOF was determined using the total Ni content obtained from XRF measurements (TOF_{XRF}) and by use of the wt% Ni determined by the charge of $\text{Ni}^{2+}/\text{Ni}^{3+}$ redox peak (TOF_Q), as previously demonstrated by Nakagawa⁴¹ and Chen⁴² (Fig. 6d). As shown in Fig. 6d, whilst the two methods give different absolute values of the TOF, the trend $\text{Ni-HTO} > \text{Ni-KMO} > \text{Ni-HMO}$ is the same. Interestingly, despite the results above indicating that the Ni-KMO had the larger catalytic activity per geometrical surface area, Ni-HTO displays better TOF values (Table 2), indicating that it possesses the highest active atom efficiency. Determination of the % Ni from the integrated charges and XRF measurements both strongly support our claim of low-loading of active catalysts for OER catalysis. For example, as the XRF result is overall evaluation of Ni content in catalyst, TOF calculation based on XRF display the lower limitation. 2 wt% Ni in Ni-HTO shows a TOF_{XRF} of 0.1090

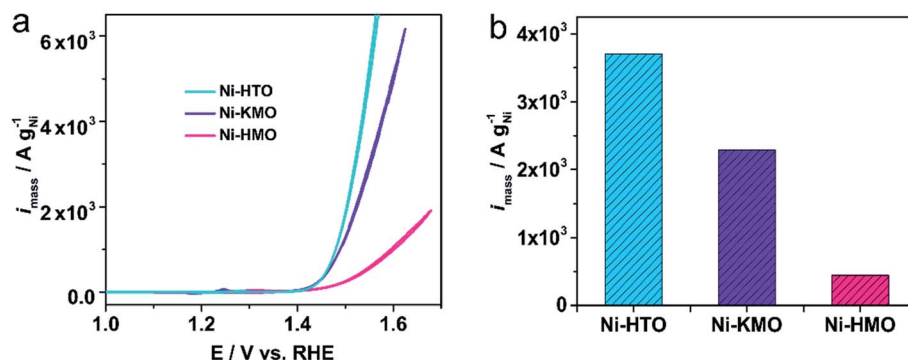


Fig. 7 (a) Voltammetric mass activity of Ni-KMO, Ni-HMO, and Ni-HTO in 1 M NaOH. Scan rate $\nu = 20 \text{ mV s}^{-1}$, (b) corresponding activity at 1.53 V vs. RHE.



$\pm 0.0067 \text{ s}^{-1}$ at 1.47 V vs. RHE. While the TOF_Q of Ni-HTO is $4.22 \pm 0.05 \text{ s}^{-1}$ based on integrated charge, which is an order of magnitude higher than TOF_{XRF} (Fig. S21†).

The mass activities of our ion-exchanged catalysts in Fig. 7 are compared with other Ni-based catalysts previously reported in the literature, including a variety of Ni structures (Ni foam, self-supported NiOOH nanocatalysts) and shown in Table S5.† Ni-KMO and Ni-HMO were found to have the largest mass activities by a factor of 4 or more. The mass activities of these 2D Ti-O and Mn-O Ni catalysts surpasses the state-of-the-art Ni-based catalysts, such as NiV-LDH ($144 \text{ g}_{\text{Ni}}^{-1}$ @300 mV)⁴³ or single Au atoms anchored on NiFe-LDH ($117 \text{ g}_{\text{Ni}}^{-1}$ @280 mV).⁴⁴

4. Conclusion

This work described the preparation and characterization of highly active OER catalysts based on the facile ion-exchange of Ni atoms into the 2D birnessite and lepidocrocite titanate structures. A direct correlation between the Ni redox peaks and the OER activity was demonstrated. After the evaluation of different catalysts as a function of the Ni content it was found that Ni-KMO and Ni-HTO show excellent mass activity of $602 \text{ A g}_{\text{Ni}}^{-1}$ and $710 \text{ A g}_{\text{Ni}}^{-1}$ at overpotential of 240 mV, respectively. Ni-HTO shows a 6-fold increase in catalytic activity per mass of Ni in comparison with the state-of-the-art Ni-based nanocatalysts ($\text{Au@Ni}_{7/8}\text{Fe}_{1/8}\text{LDH}$).⁴⁴ By evaluating the electrochemical performance at varying pHs in NaOH, it was found that the catalytic activity was enhanced at higher pH and in the presence of K cations, $\eta_{\text{onset}} = 100 \text{ mV}$ and $\eta_{10} = 206 \text{ mV}$ were obtained in comparison with those in the presence of Na cations ($\eta_{\text{onset}} = 130 \text{ mV}$, $\eta_{10} = 240 \text{ mV}$). The intercalation mechanism and oxidation state of Ni and Mn have been explored by the XAS technique. Our results demonstrate that Ni intercalation occurs at the Mn^{3+} sites by reducing Mn^{3+} to Mn^{2+} . As mentioned above, the *ex situ* XAS/XES results show that the Ni-content in each of the samples is proportional to the concentrations Mn^{3+} sites. Even though Ni is present in the +2 oxidation state for all the samples, the Ni coordination/structures obtained for Ni-KMO and Ni-HMO were different. We associate the differences in OER activity to the differences in coordination/structures between Ni-KMO and Ni-HMO.

It was also demonstrated that while the intercalated Ni is the active site for oxygen evolution reaction, Mn sites do not contribute to the OER. Thus, the evaluation of intrinsic activity, such as TOFs and mass activity, by using the Ni content offers reliable results.

In brief, highly efficient catalysts with Ni confined in 2D layered metal oxides towards OER have been synthesized by ion exchange reactions. Our synthesis protocol provides evidence for the broader applicability of our employed methodology to other layered metal oxide structures and to compositions other than manganese oxides.

Author contributions

Y. P., P. R. and L. H. conceived the experiments. All the electrochemical experiments, XRD, XRF, SEM, TGA characterization

and analyses were performed by Y. P. XANES and EXAFS characterization and analyses were performed by M. J. L. and V. C. XES measurements were collected and analyzed by V. C. and A. E. R. Low magnification HAADF- and BF-STEM, and SAD characterization were performed by L. A. J. High resolution HAADF-STEM and EDX analyses were performed by Q. W., Z. S. and M. G. All authors contributed to the analysis of the results, discussion, writing and revision of the manuscript. All authors have given approval to the final version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was financially supported by Southern University of Science and Technology (SUSTech) start fund through Shenzhen Peacock Talent program, Guangdong Innovative and Entrepreneurial Research Team Program (No. 2016ZT06N532), Guangdong Provincial Key Laboratory of Materials for Electric Power (2018B030322001) and Shenzhen Clean Energy Research Institute (CERI-KY-2019-003). P. R. acknowledges the University of Birmingham for financial support through the Birmingham fellowship program. This work was also supported by the Pico Center at SUSTech that receives support from Presidential fund and Development and Reform Commission of Shenzhen Municipality. The authors wish to acknowledge the Diamond Light Source for provision of beamtime (SP21659 and SP19850).

References

- 1 J. Kibsgaard and I. Chorkendorff, Considerations for the scaling-up of water splitting catalysts, *Nat. Energy*, 2019, **4**(6), 430–433.
- 2 C. Wei, R. R. Rao, J. Peng, B. Huang, I. E. Stephens, M. Risch, Z. J. Xu and Y. Shao-Horn, Recommended Practices and Benchmark Activity for Hydrogen and Oxygen Electrocatalysis in Water Splitting and Fuel Cells, *Adv. Mater.*, 2019, 1806296.
- 3 F. Song, L. Bai, A. Moysiadou, S. Lee, C. Hu, L. Liardet and X. Hu, Transition metal oxides as electrocatalysts for the oxygen evolution reaction in alkaline solutions: an application-inspired renaissance, *J. Am. Chem. Soc.*, 2018, **140**(25), 7748–7759.
- 4 M. P. Browne, Z. Sofer and M. Pumera, Layered and two dimensional metal oxides for electrochemical energy conversion, *Energy Environ. Sci.*, 2019, **12**(1), 41–58.
- 5 W. Zhang and K. Zhou, Ultrathin Two-Dimensional Nanostructured Materials for Highly Efficient Water Oxidation, *Small*, 2017, **13**(32), 1700806.
- 6 R. Subbaraman, D. Tripkovic, K.-C. Chang, D. Strmcnik, A. P. Paulikas, P. Hirunsit, M. Chan, J. Greeley, V. Stamenkovic and N. M. Markovic, Trends in activity for the water electrolyser reactions on 3d M (Ni, Co, Fe, Mn) hydr(oxy) oxide catalysts, *Nat. Mater.*, 2012, **11**(6), 550.



- 7 M. S. Burke, M. G. Kast, L. Trotochaud, A. M. Smith and S. W. Boettcher, Cobalt-iron (oxy) hydroxide oxygen evolution electrocatalysts: the role of structure and composition on activity, stability, and mechanism, *J. Am. Chem. Soc.*, 2015, **137**(10), 3638–3648.
- 8 Y. Zhao, C. Chang, F. Teng, Y. Zhao, G. Chen, R. Shi, G. I. Waterhouse, W. Huang and T. Zhang, Defect-Engineered Ultrathin δ -MnO₂ Nanosheet Arrays as Bifunctional Electrodes for Efficient Overall Water Splitting, *Adv. Energy Mater.*, 2017, **7**(18), 1700005.
- 9 C. Luan, G. Liu, Y. Liu, L. Yu, Y. Wang, Y. Xiao, H. Qiao, X. Dai and X. Zhang, Structure effects of 2D materials on α -nickel hydroxide for oxygen evolution reaction, *ACS Nano*, 2018, **12**(4), 3875–3885.
- 10 F. Song and X. Hu, Exfoliation of layered double hydroxides for enhanced oxygen evolution catalysis, *Nat. Commun.*, 2014, **5**, 4477.
- 11 F. Song and X. Hu, Ultrathin cobalt-manganese layered double hydroxide is an efficient oxygen evolution catalyst, *J. Am. Chem. Soc.*, 2014, **136**(47), 16481–16484.
- 12 H. Wang, J. Zhang, X. Hang, X. Zhang, J. Xie, B. Pan and Y. Xie, Half-Metallicity in Single-Layered Manganese Dioxide Nanosheets by Defect Engineering, *Angew. Chem., Int. Ed.*, 2015, **54**(4), 1195–1199.
- 13 W. Liu, H. Liu, L. Dang, H. Zhang, X. Wu, B. Yang, Z. Li, X. Zhang, L. Lei and S. Jin, Amorphous cobalt-iron hydroxide nanosheet electrocatalyst for efficient electrochemical and photo-electrochemical oxygen evolution, *Adv. Funct. Mater.*, 2017, **27**(14), 1603904.
- 14 J. Li, X. Gao, Z. Li, J. H. Wang, L. Zhu, C. Yin, Y. Wang, X. B. Li, Z. Liu and J. Zhang, Superhydrophilic graphdiyne accelerates interfacial mass/electron transportation to boost electrocatalytic and photoelectrocatalytic water oxidation activity, *Adv. Funct. Mater.*, 2019, **29**(16), 1808079.
- 15 V. Vij, S. Sultan, A. M. Harzandi, A. Meena, J. N. Tiwari, W.-G. Lee, T. Yoon and K. S. Kim, Nickel-based electrocatalysts for energy-related applications: oxygen reduction, oxygen evolution, and hydrogen evolution reactions, *ACS Catal.*, 2017, **7**(10), 7196–7225.
- 16 A. Bergmann, T. E. Jones, E. M. Moreno, D. Teschner, P. Chernev, M. Gliech, T. Reier, H. Dau and P. Strasser, Unified structural motifs of the catalytically active state of Co (oxyhydr) oxides during the electrochemical oxygen evolution reaction, *Nat. Catal.*, 2018, **1**(9), 711.
- 17 Y. Hou, M. Qiu, M. G. Kim, P. Liu, G. Nam, T. Zhang, X. Zhuang, B. Yang, J. Cho and M. Chen, Atomically dispersed nickel-nitrogen-sulfur species anchored on porous carbon nanosheets for efficient water oxidation, *Nat. Commun.*, 2019, **10**(1), 1392.
- 18 A. Munir, T. Haq, A. Qurashi, H. Rehman, A. Ul-Hamid and I. Hussain, Ultrasmall Ni/NiO Nanoclusters on Thiol-Functionalized and-Exfoliated Graphene Oxide Nanosheets for Durable Oxygen Evolution Reaction, *ACS Appl. Energy Mater.*, 2018, **2**(1), 363–371.
- 19 R. Ma and T. Sasaki, Nanosheets of oxides and hydroxides: ultimate 2D charge-bearing functional crystallites, *Adv. Mater.*, 2010, **22**(45), 5082–5104.
- 20 D. Deng, K. Novoselov, Q. Fu, N. Zheng, Z. Tian and X. Bao, Catalysis with two-dimensional materials and their heterostructures, *Nat. Nanotechnol.*, 2016, **11**(3), 218.
- 21 T. Takashima, K. Hashimoto and R. Nakamura, Mechanisms of pH-dependent activity for water oxidation to molecular oxygen by MnO₂ electrocatalysts, *J. Am. Chem. Soc.*, 2012, **134**(3), 1519–1527.
- 22 O. Diaz-Morales, D. Ferrus-Suspedra and M. T. Koper, The importance of nickel oxyhydroxide deprotonation on its activity towards electrochemical water oxidation, *Chem. Sci.*, 2016, **7**(4), 2639–2645.
- 23 J. E. Post and D. R. Veblen, Crystal structure determinations of synthetic sodium, magnesium, and potassium birnessite using TEM and the Rietveld method, *Am. Mineral.*, 1990, **75**(5–6), 477–489.
- 24 J. C. Conesa, Electronic structure of the (undoped and Fe-doped) NiOOH O₂ evolution electrocatalyst, *J. Phys. Chem. C*, 2016, **120**(34), 18999–19010.
- 25 M. Belli, A. Scafati, A. Bianconi, S. Mobilio, L. Palladino, A. Reale and E. Burattini, X-ray absorption near edge structures (XANES) in simple and complex Mn compounds, *Solid State Commun.*, 1980, **35**(4), 355–361.
- 26 P. Ghigna, G. Flor and G. Spinolo, An Mn-K Edge XAS Investigation on the Crystal Chemistry of Cd₁- δ Mn₂O_y, *J. Solid State Chem.*, 2000, **149**(2), 252–255.
- 27 J.-K. Chang, M.-T. Lee and W.-T. Tsai, *In situ* Mn K-edge X-ray absorption spectroscopic studies of anodically deposited manganese oxide with relevance to supercapacitor applications, *J. Power Sources*, 2007, **166**(2), 590–594.
- 28 D. Friebe, M. W. Louie, M. Bajdich, K. E. Sanwald, Y. Cai, A. M. Wise, M.-J. Cheng, D. Sokaras, T.-C. Weng and R. Alonso-Mori, Identification of highly active Fe sites in (Ni, Fe) OOH for electrocatalytic water splitting, *J. Am. Chem. Soc.*, 2015, **137**(3), 1305–1313.
- 29 M. Görlin, P. Chernev, J. Ferreira de Araújo, T. Reier, S. Dresch, B. Paul, R. Krähnert, H. Dau and P. Strasser, Oxygen evolution reaction dynamics, faradaic charge efficiency, and the active metal redox states of Ni-Fe oxide water splitting electrocatalysts, *J. Am. Chem. Soc.*, 2016, **138**(17), 5603–5614.
- 30 M. Görlin, J. Ferreira de Araújo, H. Schmies, D. Bernsmeier, S. Dresch, M. Gliech, Z. Jusys, P. Chernev, R. Krähnert, H. Dau and P. Strasser, Tracking Catalyst Redox States and Reaction Dynamics in Ni-Fe Oxyhydroxide Oxygen Evolution Reaction Electrocatalysts: The Role of Catalyst Support and Electrolyte pH, *J. Am. Chem. Soc.*, 2017, **139**(5), 2070–2082.
- 31 B. S. Yeo and A. T. Bell, *In situ* Raman study of nickel oxide and gold-supported nickel oxide catalysts for the electrochemical evolution of oxygen, *J. Phys. Chem. C*, 2012, **116**(15), 8394–8400.
- 32 S. R. Mellsop, A. Gardiner, B. Johannessen and A. T. Marshall, Structure and transformation of oxyhydroxide films on Ni anodes below and above the oxygen evolution potential in alkaline electrolytes, *Electrochim. Acta*, 2015, **168**, 356–364.



- 33 K. Xu, P. Chen, X. Li, Y. Tong, H. Ding, X. Wu, W. Chu, Z. Peng, C. Wu and Y. Xie, Metallic nickel nitride nanosheets realizing enhanced electrochemical water oxidation, *J. Am. Chem. Soc.*, 2015, **137**(12), 4119–4125.
- 34 J. Liu, Y. Zheng, Y. Jiao, Z. Wang, Z. Lu, A. Vasileff and S. Z. Qiao, NiO as a bifunctional promoter for RuO₂ toward superior overall water splitting, *Small*, 2018, **14**(16), 1704073.
- 35 C. Stoffelsma, P. Rodriguez, G. Garcia, N. Garcia-Araez, D. Strmcnik, N. M. Marković and M. T. Koper, Promotion of the oxidation of carbon monoxide at stepped platinum single-crystal electrodes in alkaline media by lithium and beryllium cations, *J. Am. Chem. Soc.*, 2010, **132**(45), 16127–16133.
- 36 G. García, C. Stoffelsma, P. Rodriguez and M. T. Koper, Influence of beryllium cations on the electrochemical oxidation of methanol on stepped platinum surfaces in alkaline solution, *Surf. Sci.*, 2015, **631**, 267–271.
- 37 J. Suntivich, E. E. Perry, H. A. Gasteiger and Y. Shao-Horn, The Influence of the Cation on the Oxygen Reduction and Evolution Activities of Oxide Surfaces in Alkaline Electrolyte, *Electrocatalysis*, 2013, **4**(1), 49–55.
- 38 M. Xiao, Y. Tian, Y. Yan, K. Feng and Y. Miao, Electrodeposition of Ni(OH)₂/NiOOH in the presence of urea for the improved oxygen evolution, *Electrochim. Acta*, 2015, **164**, 196–202.
- 39 W. Jin, H. Du, S. Zheng, H. Xu and Y. Zhang, Comparison of the oxygen reduction reaction between NaOH and KOH solutions on a Pt electrode: the electrolyte-dependent effect, *J. Phys. Chem. B*, 2010, **114**(19), 6542–6548.
- 40 A. C. Thenuwara, E. B. Cerkez, S. L. Shumlas, N. H. Attanayake, I. G. McKendry, L. Frazer, E. Borguet, Q. Kang, R. C. Remsing, M. L. Klein, M. J. Zdilla and D. R. Strongin, *Angew. Chem., Int. Ed.*, 2016, **55**(35), 10381–10385.
- 41 T. Nakagawa, C. A. Beasley and R. W. Murray, Efficient electro-oxidation of water near its reversible potential by a mesoporous IrO_x nanoparticle film, *J. Phys. Chem. C*, 2009, **113**(30), 12958–12961.
- 42 J. Y. Chen, J. T. Miller, J. B. Gerken and S. S. Stahl, Inverse spinel NiFeAlO₄ as a highly active oxygen evolution electrocatalyst: promotion of activity by a redox-inert metal ion, *Energy Environ. Sci.*, 2014, **7**(4), 1382–1386.
- 43 K. Fan, H. Chen, Y. Ji, H. Huang, P. M. Claesson, Q. Daniel, B. Philippe, H. Rensmo, F. Li, Y. Luo and L. Sun, Nickel–vanadium monolayer double hydroxide for efficient electrochemical water oxidation, *Nat. Commun.*, 2016, **7**(1), 11981.
- 44 J. Zhang, J. Liu, L. Xi, Y. Yu, N. Chen, S. Sun, W. Wang, K. M. Lange and B. Zhang, Single-Atom Au/NiFe Layered Double Hydroxide Electrocatalyst: Probing the Origin of Activity for Oxygen Evolution Reaction, *J. Am. Chem. Soc.*, 2018, **140**(11), 3876–3879.

