Degradation induced lattice anchoring self-passivation in CsPbl₃₋ₓBrₓ †

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The all-inorganic halide perovskite (CsPbl₃) holds promise for photovoltaic applications but suffers from a detrimental phase transformation to a non-perovskite phase δ-CsPbl₃ at low-temperature. Of the different perovskite polymorphs, there has been a wide range of studies on γ-CsPbl₃ due to its kinetic stability at near room-temperature. However, synthesis routes to this and other all-inorganic halide perovskites are still not ideal, requiring uneconomical elimination of humidity as well as quenching from elevated temperature. Water/moisture is commonly meticulously avoided due the fact that it can accelerate the detrimental degradation of the perovskite. In our synthesis, we used an alternative approach of engineering an in situ degradation process to form a dual-functional PbI(OH) protective covering and succeeded in performing the first room-temperature synthesis of γ-CsPbl₃ under ambient humidity. The vastly improved stability benefits from both lattice anchoring and physical coverage of γ-CsPbl₃ by an ultra-thin PbI(OH) layer. The resultant γ-CsPbl₃ is stable for more than 2 months under ambient conditions (25 °C, RH = 30–60%) and more than 12 hours at 175 °C in air without any degradation. Furthermore, we show that this novel facile method can be successfully applied to mixed halide perovskites such as CsPbl₂Br, and this has allowed the first experimental synthesis of the γ-polymorph of CsPbl₂Br. Thus, our work provides an efficient degradation-induced lattice-anchoring self-stabilization strategy and a new avenue to the economical synthesis of all-inorganic perovskite materials at room-temperature under ambient conditions.

Introduction

Organic–inorganic halide perovskites have attracted tremendous attention in the past decade because of their good photovoltaic performance. However, the thermal instability resulting from the volatile nature of the organic cations (e.g., CH₃NH₃⁺) impedes their development toward commercialization. All-inorganic perovskites, realized by replacing the unstable organic cations with inorganic Cs⁺, are promising candidates for more stable perovskite solar cells. Unfortunately, α-CsPbl₃ is only stable above 325 °C because of the structural instability of the perovskite resulting from the smaller size of Cs⁺ cations. It is first converted into the low-temperature β/γ perovskite phase and then finally to non-perovskite δ-CsPbl₃ at room-temperature as shown in Fig. 1a. This phase instability of CsPbl₃ has attracted considerable interest in the development of strategies to achieve the formation of these low-temperature photoactive β/γ perovskite phases, particularly the γ phase.

Single crystals of γ-CsPbl₃ have been grown via a solid-state method through quenching with strict control over humidity. Thin-films of γ-CsPbl₃ have also been widely studied in recent years. In order to reduce the synthesis temperature, HI has been introduced for the preparation of the γ-CsPbl₃ film so as to decrease the annealing temperature for γ-CsPbl₃ to 100 °C. More recently, co-evaporation of CsI and PbI₂ onto the substrate has allowed a lower temperature annealing of 50 °C. However, the preparation of γ-CsPbl₃ at room-temperature, especially in the ambient environment (with its associated humidity challenges), has not been reported previously. To this end, we have been examining the possibility of synthesizing this phase via a solution route. Such solution-processed perovskites have various merits such as the potential commercial benefits of low-cost room-temperature preparation and scale-up. Nevertheless, this approach is not without its challenges, especially for the preparation of γ-CsPbl₃ as a result of the greater inherent thermodynamic stability of the undesirable non-perovskite δ-CsPbl₃ product. In particular, the presence of moisture or heat is known to accelerate the phase transformation into δ-CsPbl₃. Various strategies, such as reducing the grain size...
and composition engineering, have been examined to improve the stability. However, these routes entail their own issues, such as the introduction of potentially undesirable defects and the consequent influence on the band gap of the perovskite. Other potential strategies include surface passivation strategies, such as via ligand passivation, which show some promise, although the uniformity and compactness of any resultant passivation layer remains a challenge to be addressed.

In view of these challenges, our aim was to find a feasible solution-processing approach to prepare this all-inorganic perovskite at room-temperature in air in order to kick-start future advanced applications of this material (Fig. 1a). In particular we aimed to address the issue of relying on uneconomical dry synthesis environments, by designing a surface degradation process that would lead to uniform growth of a surface passivation layer, which would be expected to provide a high-quality surface protective covering while preventing uncontrolled degradation throughout the particles. In our thoughts behind this strategy, we were drawn to the degradation reaction of $\gamma$-CsPbI$_3$ in water, which is a universal problem for perovskite and has led researchers to naturally avoid its presence for the preparation of these perovskite phases. Our approach was to try to utilize this process to provide a surface degradation layer to passivate against further degradation. The challenge is to achieve a uniform passivation layer, whose lattice matches that of the perovskite in order to prevent delamination and so creating a fresh surface. As a common example, iron readily rusts, due to the fact that the corrosion product delaminates, creating a fresh surface, while Al is kinetically stabilized by a surface layer of the oxide which protects the metal from further corrosion. The aim of stabilizing $\gamma$-CsPbI$_3$ therefore is to create a lattice matched surface decomposition product to provide a similar protective layer. In particular, it is known that water can degrade these perovskite systems, and so we explored solution routes containing water to examine whether we could induce and then cap this degradation process to form a stable protective layer.

Here we report this new solution processing preparation route to stable $\gamma$-CsPbI$_3$ at room-temperature in an ambient environment. By controlling the degradation process of perovskite with the presence of water, an ultra-thin single-crystal PbI(OH) epitaxial shell is grown on the surface of $\gamma$-CsPbI$_3$ micro-crystals assisted by the presence of acetate and PVP. This in situ produced PbI(OH) passivation shell provides a dense protective covering for the perovskite, resulting for the first time in both excellent thermal stability and moisture stability of $\gamma$-CsPbI$_3$ in air. This method has also been successfully applied to CsPbI$_2$Br and has allowed the preparation of the low-temperature $\gamma$-CsPbI$_2$Br phase for the first time.

### Experimental

#### Materials

Anhydrous isopropanol (99.8%) was purchased from Acros Organics. Methanol (99.8%) and ethanol (99.8%) were purchased from Alfa Aesar. Lead acetate trihydrate (99.995%) and cesium iodide (CsI, 99.995%) were purchased from Alfa Aesar. PVP 4000 was purchased from Sigma Aldrich. All materials were used as received.

#### Synthesis

$\gamma$-CsPbI$_3$/CsPbI$_3@PbI(OH)$. 2 mmol cesium iodide was dissolved in 10 ml methanol/water (vol 9 : 1). 0.5 mmol lead acetate trihydrate was dissolved in 10 ml methanol. For the preparation...
of CsPbI$_3$@PbI(OH), 400 µl Pb$^{2+}$ precursor solution was added into a vial with 4 ml ethanol. The ethanol was used as an anti-solvent, in order to accelerate the precipitation of the perovskite by changing the polarity of the solvent system. The absence of ethanol leads to rapid degradation of the black phase in a few seconds owing to the strong polarity of water and methanol. γ-CsPbI$_3$ was obtained when 300 µl Cs$^+$ precursor solution was injected. Stable CsPbI$_3$@PbI(OH) was obtained by adding 200 mg PVP 4000 closely following the formation of γ-CsPbI$_3$. The CsPbI$_3$@PbI(OH) was washed with ultra-dry isopropanol (IPA) and dried in air naturally. Note: the ultra-dry solvent is not required during the synthesis and is only used for washing after synthesis.

γ-CsPbI$_3$Br/CsPbI$_3$Br@PbI(OH). CsPbI$_3$Br was synthesized by substituting a 1/3 mole ratio of CsI with CsBr. CsPbI$_3$Br@PbI(OH) was then obtained with the same method as that for the preparation of CsPbI$_3$@PbI(OH).

### Results and discussion

#### Characterization of γ-CsPbI$_3$

Through our new approach, γ-CsPbI$_3$ is directly prepared via a solution route at room-temperature in an ambient environment even in the presence of high humidity, bypassing the previous need to quench high-temperature z-CsPbI$_3$ to obtain this phase. The perovskite is then further self-stabilized by surface degradation to prevent conversion into the undesired non-perovskite phase δ-CsPbI$_3$ (Fig. 1a). When the Cs$^+$ and Pb$^{2+}$ precursor solutions are added into ethanol, a black colloid appears immediately through a simple one-step reaction as shown in Fig. 1b and eqn (1).

$$3\text{CsI} + \text{Pb(CH}_3\text{COO)}_2 \rightarrow \gamma-\text{CsPbI}_3 + 2\text{Cs(CH}_3\text{COO)} \quad (1)$$

As in the prior method of rapid quenching from high temperature to “freeze in” the desired γ-CsPbI$_3$,$^{14}$ in our method, a similar rule is obeyed in that a high reactant concentration and a rapid precipitation process, which is completed in seconds, are required to produce high quality γ-CsPbI$_3$ (Fig. 1c) while a low concentration and slow reaction speed results in significant amounts of the undesired δ-CsPbI$_3$ phase. This is further illustrated by the slow growth of yellow phase δ-CsPbI$_3$ from the residual reactants over a few hours after the rapid formation of the black phase γ-CsPbI$_3$ perovskite in our experiment (needle-like δ-CsPbI$_3$ can be seen to appear over time as shown in Fig. S1†). The novelty of our approach is that the reaction is performed both at room temperature and in the presence of water, with this route leading to the growth of stable CsPbI$_3$@PbI(OH) protective shell. This is illustrated by the fact that performing the synthesis of CsPbI$_3$ without adding PVP, leads to a product that turns yellow (conversion to δ-CsPbI$_3$) when dried in air. Nevertheless, when formed in the presence of PVP and separated from solution, the CsPbI$_3$@PbI(OH) remains stable under ambient conditions even after removal of the PVP by washing with isopropanol (see later). It has been reported that the C=O group in PVP interacts with the CsPbI$_3$ surface.$^{23}$ Therefore, there is expected to be an initial chemical interaction...
between the fresh CsPbI₃ surface and PVP as it is added into the reaction. As the perovskite surface degrades due to interaction with water into PbI(OH), this initial interaction will be broken. Instead, the PVP will be absorbed again on the PbI(OH) layer coordinated with the OH group, which might be expected to be a stronger interaction and so it will be hard to totally remove by the washing process. The role of this surface absorbed PVP is mostly likely to be limiting the speed of the degradation process and helping to contribute to the uniformity of PbI(OH) shell.

In order to confirm the formation of a protective shell of PbI(OH), we carried out high-resolution transmission electron microscope (HRTEM) measurements to characterize this shell. The results showed that the shell is uniform and forms a continuous surface as shown in Fig. 2b. HRTEM images of bulk perovskite shows the separation of the (040) planes with a spacing of 3.1 Å (Fig. 2c). In the PbI(OH) shell, the HRTEM image shows the separation of the similarly aligned (103) planes with a lattice spacing of 3.2 Å, leading to a small size mismatch of about 2.9% with the perovskite phase, and thus suggesting epitaxial growth of the degradation layer. Representations of these lattice planes are shown in Fig. S3 illustrating the good match between the lattices. While there is good crystallinity of the PbI(OH) shell on the end axes of the CsPbI₃ microrods, the PbI(OH) layer on the end surfaces of these microrods does not show good crystallinity, which may indicate less effective lattice matching here (Fig. 2d). The growth of the PbI(OH) layer was further investigated by energy dispersive X-ray (EDX) spectroscopy using a scanning transmission electron microscope (STEM). The PbI(OH) layer can be distinguished by STEM-EDX mapping of the Pb element because of the different Pb densities in PbI(OH) compared to CsPbI₃ as shown in Fig. 2e. The stacked mapping of the elements O and I reveals an O-rich area at the edge of the crystal, corresponding to the PbI(OH) shell. The CsPbI₃@PbI(OH) particles were further characterized by X-ray photo-electron spectroscopy (XPS) measurements. The XPS signal at a bonding energy of 531.15 eV, corresponding to the featured peak of O in a metal hydroxide, is assigned to PbI(OH).

Stability of γ-CsPbI₃

In order to probe the stability in air of these CsPbI₃@PbI(OH) particles and confirm the protective nature of the PbI(OH) shell, the particles were washed with ultra-dry isopropanol three times in order to remove any absorbed PVP. To confirm PVP removal, Fourier Transform Infrared Spectra (FTIR) were recorded, which shows that the PVP is mostly removed after washing 3 times (see Fig. S4†). Significantly, even after removal of the PVP, the PXRD pattern shows that CsPbI₃@PbI(OH) doesn’t show any degradation after keeping it in ambient air (RH = 30–60%) for more than 2 months as shown in Fig. 3a. Furthermore, it shows no degradation after heating at 175 °C for 12 h in air with an environmental relative humidity of 60% as shown in Fig. 3b. At higher temperatures (200 °C), it does transform into δ-CsPbI₃, which can be attributed to the dehydration and hence decomposition of the PbI(OH) surface layer. To further prove its excellent stability, we recorded the digital photos, SEM images and UV-vis transmittance spectra of the sample before and after heating in air for 12 h. These showed that no obvious changes triggered by degradation can be observed as shown in Fig. S5 and S6.† This heating experiment, along with the previous results for samples where the PVP
has been washed away, support the fact that the origin of the improved stability is due to the PbI(OH) shell, rather than other factors, such as PVP coverage. Rather the PVP is present to cap the initial degradation process and ensure that a uniform surface covering is obtained. Thus, our results show that this new synthesis method is efficient in the stabilization of micron-sized particles of γ-CsPbI₃ towards both moisture and heating.

We have further studied the tolerance of CsPbI₃@PbI(OH) to the amount of water in solution and remarkably found that the perovskite is stable with a water concentration as high as 29%. In these experiments, 400 μl, 800 μl, 1.6 ml, and 2 ml water were added to γ-CsPbI₃ in 2 ml mother liquor (containing PVP and cesium acetate). The volume concentrations of water in the above solutions are calculated to be 16%, 29%, 44%, and 50%. Fig. 3c shows the PXRD patterns of samples after soaking in the above solutions for 12 h at room temperature. Remarkably, the perovskite is stable under these conditions in a water concentration up to 29%. At higher water concentrations, peaks belonging to PbI₂ begin to appear, indicating the decomposition of the PbI(OH) shell in such high water concentrations, with γ-CsPbI₃ completely decomposed in the 50% water solution. This high tolerance of CsPbI₃@PbI(OH) to water explains why our synthesis can be readily conducted in its presence.

In order to demonstrate the promise of this new method for scalable preparation of stable perovskite semiconductors in solution, we successfully scaled up to 100 ml volume as shown in Fig. S5.† We then considered whether the method could be successfully applied for the synthesis of other all-inorganic perovskites, such as CsPbI₂Br.

**Synthesis and characterization of γ-CsPbI₂Br**

CsPbI₂Br has attracted considerable interest worldwide because it has been reported to show improved stability by partial substitution of I⁻ with the smaller Br⁻ anion. Prior studies have shown the synthesis of photoactive α-CsPbI₂Br through heating followed by quenching to room temperature to kinetically stabilise this phase. Nevertheless, the low-temperature β/γ-CsPbI₂Br polymorphs are also expected to show similar properties to α-CsPbI₂Br, but their synthesis has proved elusive.26

Our method has the merit of preparation at room-temperature and so omits the need to prepare the high-temperature phase. Thus, through our room temperature solution method, we are able to prepare the low temperature γ-CsPbI₂Br perovskite polymorph for the first time. Fig. 4a shows photographs of the CsPbI₃@PbI(OH) and CsPbI₂Br@PbI(OH) colloidal solutions. The reddish colour of CsPbI₂Br@PbI(OH) confirms the introduction of Br⁻ into the lattice. XRD data for this phase were collected and structural refinement confirmed this phase to be γ-CsPbI₂Br with lattice parameters \( a = 8.598(1) \) Å, \( b = 12.224(2) \) Å, and \( c = 8.496(1) \) Å (Fig. S6†). Interestingly these structural studies showed the presence of some I/Br ordering, with the Br located on the I2 site (see ESI Table 1†). Furthermore, TEM images confirm a similar core–shell structure for CsPbI₂Br@PbI(OH) as shown in Fig. 4b. As observed for CsPbI₃, CsPbI₂Br was degraded into the yellow non-perovskite phase.
rapidly in samples when there was no PbI(OH) protective layer, as shown in Fig. 4c.

Photo-luminescence (PL) spectra have been recorded for CsPbI@PbI(OH) and CsPbI–Br@PbI(OH) as shown in Fig. 4d. CsPbI@PbI(OH) exhibits an emission peak at 705 nm, which is consistent with previous results at 709 nm.13 CsPbI–Br@PbI(OH) exhibits a band edge emission at 655 nm, similar to that reported for the α polymorph of CsPbI2Br.3,27 These results show that the ultra-thin PbI(OH) shell does not have a significant effect on the photo-electronic properties of the perovskite.

Conclusions

In summary, we have shown a facile solution approach to prepare γ-CsPbI3 micro-crystals in an ambient environment for the first time. The rapid precipitation process of γ-CsPbI3 ensures the production of the desired perovskite phase at room-temperature, which is then stabilized by the spontaneous degradation of the perovskite in water, which leads to the epitaxial growth of an ultra-thin and single-crystal PbI(OH) shell on the perovskite surface. The acetate anion is believed to play a key role in controlling the degradation of perovskite to ensure formation of PbI(OH) while PVP contributes to capping the process and ensuring the PbIOH shell compactness and uniformity. These coated γ-CsPbI3 microcrystals exhibit vastly improved stability both towards moisture and heating in air, which is attributed to the lattice anchoring and uniform physical coverage of the PbI(OH) shell. Our method has also been successfully extended to other inorganic perovskites with the synthesis of CsPbI2Br. Benefiting from our room-temperature synthesis, CsPbI2Br forms the low-temperature γ perovskite phase, which represents the first experimental report of γ-CsPbI2Br. Uniquely, our study provides a new avenue for the synthesis of stable solar perovskite materials, making use of degradation processes to form a protective stabilization layer on the perovskite. It shows that the bane of many a solar perovskite synthesis (water) can also be a boon, providing that, as in the case here, the degradation products can be tailored to induce surface stability.

Author contributions

J. X., Z. H., and P. R. S. conceived the idea and wrote the manuscript. J. X., B. D., E. D., and A. M. collected the XRD data and performed the structural refinements. J. X. synthesised all the materials and performed the initial characterisation measurements and stability studies. X. F., Z. Z., and S. C. contributed to the TEM and EDX measurements. Z. D, Y. Z., and Z. T. contributed to the data analysis. Z. H. and P. R. S. supervised this project. All the authors discussed the results.

Conflicts of interest

There are no conflicts to declare.

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