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One-step synthesis of polyethyleneimine-coated magnetite nanoparticles and their structural, magnetic and power absorption study

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Magnetic nanoparticles (NPs) are especially interesting for several biomedical applications due to their chemical surface, especially for targeted cancer imaging and therapeutics. In order to realize these applications, it is important to know their magnetic properties among other complementary properties that help to improve the understanding of the synthesis process. In this work, we report the magnetic properties of polyethyleneimine-coated magnetite (PEI-Fe₃O₄) NPs synthesized by a one-step method via the co-precipitation method and using PEI as a stabilizer. Transmission electron microscopy (TEM) images revealed agglomerated magnetic nanoparticles with an average size of ~10 nm; meanwhile, the X-ray diffraction (DRX) analysis confirmed a pure magnetite phase. The study of magnetic properties shows a superparamagnetic system with coexistence of non-interacting single NPs with a low blocking temperature (~35 K) and interacting NPs in the aggregates with a higher blocking temperature (>150 K), in which the interparticle interactions of magnetic cores dominate over surface spin disorder. The interaction between the surface spin-disorder layer and NP core was found to be weak, related to a weak exchange bias effect. A maximum specific loss power (SLP) value of 70 W g⁻¹ was obtained ($f = 571$ kHz and $H = 23.87$ kA m⁻¹) indicating that the magnetic response plays a crucial role in determining the heating efficiency for future applications.

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1. Introduction

Iron oxide magnetic NPs with an appropriate biocompatible coating to improve colloidal stability¹ and a good biocompatibility or low cytotoxicity are increasingly used in many biomedical applications including protein separation,² drug and gene delivery,³ magnetic resonance imaging (MRI)⁴ and magnetic hyperthermia therapy.^{5,6}

Magnetic hyperthermia is an alternative to therapeutic treatments which uses hysteresis cycle of magnetic nanoparticles (MNPs) to locally increase cancer cells temperature to values of 43–45 °C inducing their apoptosis or death.⁷ The heating capacity of MNPs with colloidal stability under an alternating magnetic field (AMF) is quantified by the specific loss power (SLP).⁸ In order to improve the heating process, it is important to know the influence of the different parameters such as size and shape of NPs, solvent, colloidal stability, NPs biocompatibility, and intrinsic magnetic properties. Those

parameters can lead to a more complex behavior of the system that could affect the heating efficiency of the NPs.

Also, it is important to consider the influence of the surface, interface effects such as surface spin disorder, defects, breaking of exchange bonds, changes in the surface atom coordination number,⁹ and also the distance among MNPs assemblies that leads to interparticle interactions¹⁰ that can dominate over single domain nanoparticles response and affect the heating efficiency.

Some nanoparticulated features such as the surface spin disorder and magnetic frustrations, related to the presence of structurally disordered grain boundaries, can provide a dominant contribution to the effective anisotropy and lead to a surface spin-glass like state at low temperatures.¹¹ It is known that the occurrence of a spin-glass state is related to a magnetic disorder, randomness (exchange, anisotropy, field), and frustrations,¹² where the competing interactions (ferromagnetic and antiferromagnetic) show equivalent strengths.

In addition, in a system where interparticle interactions are non-negligible, the system eventually shows collective behavior, which overcomes the anisotropy properties of individual particles, and leads to the increase of the blocking temperature (T_B). When the interparticle interactions are strong enough in a nanoparticle ensemble,¹³ also can lead to the spin-glass behavior besides the increment of the T_B .¹⁴

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In particle systems, collective effects related to different kinds of magnetic interactions play an important role such as the long-range dipole–dipole interactions that are the predominant mechanism; meanwhile, the short-range exchange interactions play a leading role in nanoparticle assemblies, where the electrons at the surface of the particles are in close contact¹² and both mechanisms may exist simultaneously. De Toro *et al.* reported that despite the close contact of the maghemite particles (with nonmagnetic shell thickness < 3 nm) the superexchange interactions play a minor role in establishing the collective, and superspin-glass state of the NPs below a critical temperature in comparison to dipolar interactions.¹⁵ The origin of the superspin-glass behavior is strongly related to the interplay of intra- and interparticle interactions effects, where the role of dipolar interactions is very important for establishing the superspin-glass phase.¹⁴

Magnetic NPs systems can exhibit the so-called exchange bias (EB) effect, which is related to the exchange coupling between core and surface spins at the interface and the interparticle exchange coupling. A study of NPs with Fe/Fe oxide core (10 nm)/shell (3.5 nm) structure shows a superparamagnetic behavior and the structural disorder in particles outer shell could lead to a larger number of uncompensated spins at the interface of the core–shell structure, which, in turn, causes a high magnetocrystalline anisotropy and an enhanced EB effect (22 kA m⁻¹ at 5 K and using a field of 1591.5 kA m⁻¹ to cool the system).¹⁶ On the other hand, it has been reported an EB effect for a core–shell structure of Fe₃O₄/γ-Fe₂O₃ NPs of 12 nm size with an EB field of 11.14 kA m⁻¹ at 10 K and after field cooling with 37.8 kA m⁻¹ (ref. 17) the authors reported that there is an intrinsic single-particle property, as the large magnetic anisotropy of the γ-Fe₂O₃ shell with a spin glass-like behavior or a possibly disordered magnetic state no related to the interparticle interaction.¹⁸ Meanwhile, the observed EB effect in ultrasmall ~2 nm MnFe₂O₄ NPs has been assigned to the exchange coupling between core and surface spins at the interface and the interparticle exchange coupling.¹⁶ In magnetite NPs of 40 nm a spin-glass like behavior with a freezing temperature of ~35 K was determined and could be observed in both, the in-phase and the out-phase magnetic susceptibility curves.¹⁹ These characteristics can be tuned during the synthesis process.

So far, various experimental methods have been employed to produce PEI coated magnetic NPs such as solvothermal, hydrothermal, and co-precipitation methods. They have been applied for the synthesis at high temperatures,²⁰ during long periods of synthesis⁶ using several reaction steps,^{21,22} such as first core synthesis, and after coated with PEI.^{23–27} PEI molecule is considered a good candidate for the functionalization of other molecules, the biocompatibility and stability of MNPs, that are important in the fields of biomedicine.²⁸

In order to improve and simplify the synthesis for the formation of PEI-coated MNPs, we propose an effective procedure that combines co-precipitation steps in a one-step procedure in an aqueous medium. The design of PEI-MNPs has a huge importance in the heat generation *via* electromagnetic energy conversion.

In this work, we report a simple route for an efficient and facile one-step PEI-Fe₃O₄ NPs synthesis, using the co-precipitation method. We obtained NPs of ~9.4 nm in size, and the study of their magnetic properties and the power absorption response are presented and discussed. The phase, crystal structure, and magnetic properties were characterized by X-ray diffraction, transmission electron microscopy, dc, and ac magnetic measurement and power absorption for future applications in magnetic hyperthermia.

2. Experimental section

2.1 Chemicals

Iron(III) chloride hexahydrate (FeCl₃·6H₂O), iron(II) chloride tetrahydrate (FeCl₂·4H₂O), sodium hydroxide (NaOH) and polyethylenimine branched (PEI) with a molecular weight of ~25 000 were purchased from Sigma Aldrich.

2.2 Synthesis of polyethyleneimine coated magnetite nanoparticles

The synthesis protocol used for the samples was based on a co-precipitation method. At the beginning of the synthesis, 8 mol L⁻¹ NaOH solution (water) was heated up to 90 °C by vigorous constant stirring and bubbled with N₂. After 10 min, the mixture of 1 mol L⁻¹ FeCl₂ and 2 mol L⁻¹ FeCl₃ with 2.1 g PEI (25 kDa) was dissolved in distilled water and was added in a basic solution while constantly stirring. Then the system was maintained at 90 °C for 2 h under N₂. When the precipitation was completed, the suspension with black precipitate was removed from the heating source and cooled at room temperature with an ice bath then washed with water several times in order to isolate the supernatants by magnetic decantation, and re-dispersed in water until getting pH 7, the particle presented a good water dispersibility and stability. The final product was dried with N₂ flux.

2.3 Characterization

The morphology and structure of the resulting NPs were analyzed by a high resolution transmission electron microscopy (HRTEM), using an FEI Tecnai F30 microscope operated at an acceleration voltage of 300 kV. Lattice fringes were measured from the fast-Fourier transform of HRTEM images analyses, using Gatan Digital Micrograph software. The mean particle size, and its polydispersion index, σ , were obtained by calculating the average number manually measuring the equivalent diameter of $N > 500$ particles from TEM micrographs. XRD patterns were obtained using a Rigaku Miniflex 600 diffractometer operating at 30 mA and 40 kV from 20 to 80° (2 θ value) using Cu K- α radiation (0.15418 nm). The samples were prepared placing a concentrated NPs suspension drop on a zero-diffraction silicon wafer. The Rietveld method analysis was used to confirm the structural analysis of NPs. The lattice parameters were determined using the GSAS (General Structure Analysis System) refinement. The magnetic properties of the sample were measured by ac and dc magnetic susceptibility measurements, performed on a Superconducting Quantum



Interference Device (SQUID) magnetometer, model MPMS3 (Quantum Design). The magnetization curves were recorded using a maximum applied field of 5570.4 kA m^{-1} at temperatures of 5 and 300 K. For zero-field-cooled (ZFC) and field-cooled (FC) curves were measured at temperatures between 2 to 300 K, with a cooling field $H_{\text{FC}} = 2.39 \text{ kA m}^{-1}$ (30 Oe). For ac magnetic susceptibility measurements, the data was obtained in a temperature range of 2 to 300 K, and frequencies ranging from 0.2 to 1 kHz and under an excitation field $H_{\text{ac}} = 400 \text{ A m}^{-1}$. The Specific Loss Power (SLP) of the SLP NPs was measured under AMFs as a function of the field amplitude ($13.53 \leq H_0 \leq 23.87 \text{ kA m}^{-1}$) and a fixed frequency of $f = 571 \text{ kHz}$. At 478 nm, the wavelength was used to measure the absorbance of NPs and for determining the iron concentration.

3. Results and discussion

3.1 Structural and morphological characterization

In Fig. 1 is shown the ray-X diffraction data analysis of the PEI- Fe_3O_4 NPs. The XRD pattern was refined using the Rietveld refinement method, which indicates that all diffraction peaks correspond to the cubic spinel structure of magnetite (space group: $Fd\bar{3}m$). The lattice parameter calculated from the XRD pattern is $\approx 8.37 \text{ \AA}$, which is very close to the standard lattice parameter of bulk magnetite (8.39 \AA). Additionally, the full width at half-maximum of the peaks was used to estimate the crystallite size (D_{XRD}) from the Scherrer equation. From this analysis, the calculated value was $D_{\text{XRD}} \approx 10.1 \text{ nm}$.

Fig. 2 shows the TEM images of PEI-coated Fe_3O_4 NPs, which were used to determine the structure, morphology, and size distribution. After counting $N = 518$ sizes, the histogram built using the Sturges criterion could be fitted with a log-normal distribution, yielding an average size of $\langle d \rangle = 9.6 \text{ nm}$ and a polydispersity index $\sigma = 0.19 \text{ nm}$ (see Fig. 2a). As observed in the images, the NPs show mainly spherical and octahedral shapes, and also agglomeration of particles is observed that

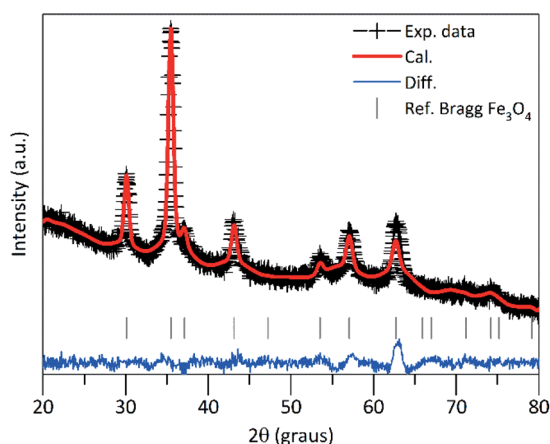


Fig. 1 X-ray diffraction pattern of the PEI- Fe_3O_4 nanoparticles. The solid red line represents the calculated pattern and the experimental data are represented by the cross symbol; the blue line at the bottom of the plot represents the difference between the experimental and calculated data. The inserted vertical lines indicate the Bragg planes.

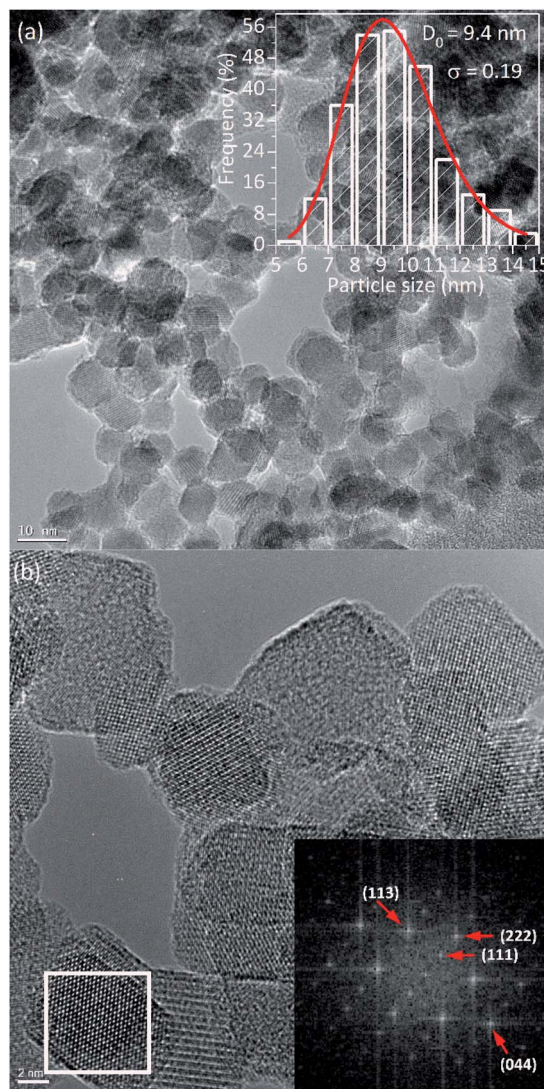


Fig. 2 (a) TEM image of the PEI- Fe_3O_4 NPs. In the inset, the histogram of particle size distribution with lognormal functions is shown. (b) The high-resolution TEM image of the NPs with the corresponding Fourier transform of the selected region is shown at the lower part.

could mimics a large particle. Especially, Fig. 2b displays the high-resolution TEM images with clear lattice fringe patterns, which confirm the high crystalline quality of the sample. The Fourier transform of the HR-TEM images show spots corresponding to the spacing distances of 4.85 \AA , 2.48 \AA , 2.59 \AA and 1.55 \AA , which are consistent with the interplanar distances of (111), (222), (113) and (044) lattices planes of magnetite, respectively.

3.2 Magnetic characterization

In order to characterize the magnetic properties in detail, the static and dynamic magnetic responses were investigated. We begin the analysis with the estimation of the T_{B} of non-interacting particles, using the average size obtained by TEM and the relation, $T_{\text{B}} = K_{\text{eff}}V/25k_{\text{B}}$, where k_{B} is the Boltzmann constant, considering the anisotropy constant (K_{eff}) of magnetite bulk. A mean value of $\langle T_{\text{B}} \rangle \approx 22 \text{ K}$ was obtained.



In Fig. 3 are shown zero-field-cooled (ZFC) and field-cooled (FC) magnetization curves of PEI-Fe₃O₄ NPs obtained with an applied field of 2.39 kA m⁻¹. The ZFC curve shows, a broad maximum temperature at $T_{\text{max}} \sim 195$ K and a shoulder in the low-temperature region at around 30 K. The high T_{max} value suggests the occurrence of strong dipolar interparticle interactions²⁹ due to the NPs being very close to each other, forming particle aggregates as observed by TEM images, which favor the interparticle interactions. Moreover, the shoulder could be related to the blocking of non-interacting single particles. At temperatures below T_B , the magnetic moments of the particles must show random orientation in a zero-field cooled condition and cannot rotate freely.³⁰ Then, the system would exhibit a transition from a high-temperature superparamagnetic state to a low temperature blocked state.

The distribution of T_B can be obtained from ZFC–FC curves according to $f(T) \approx d(M_{\text{ZFC}} - M_{\text{FC}})/dT$.^{31,32} As it is shown in the inset of Fig. 3, the experimental data shows bimodal features that suggest the occurrence of two T_B distributions, which can be modeled using the lognormal distribution. The fit provides two maxima at $\langle T_{1B} \rangle \sim 25$ K and $\langle T_{2B} \rangle \sim 105$ K. The $\langle T_{1B} \rangle$ value is close to the one expected for non-interacting single particles with the determined size from TEM images, and $\langle T_{2B} \rangle$ is not directly related to the particle size distribution but could be associated with the presence of interactions among close particles or particle agglomerations that promote the interparticle interactions. Those values obtained from the fit of the $f(T) \approx d(M_{\text{ZFC}} - M_{\text{FC}})/dT$ vs. T curve was used to simulate the ZFC and FC curves.

It is known that the effect of the particle size distribution gives rise to a distribution of T_B , which can drive to the superposition of responses coming from the superparamagnetic state (first term in eqn (1) and (2)) and in the blocked state (second term in eqn (1) and (2)).^{33,34} Then, the ZFC (eqn (1)) and FC (eqn (2)) susceptibility can be given by:

$$\chi_{\text{ZFC}} = \frac{M_S^2}{3K_{\text{eff}}} \left[\text{Ln} \left(\frac{\tau_m}{\tau_0} \right) \int_0^T \frac{T_B}{T} f(T_B) dT_B + \int_T^\infty f(T_B) dT_B \right] \quad (1)$$

$$\chi_{\text{FC}} = \frac{M_S^2}{3K_{\text{eff}}} \text{Ln} \left(\frac{\tau_m}{\tau_0} \right) \left[\frac{1}{T} \int_0^T T_B f(T_B) dT_B + \int_T^\infty f(T_B) dT_B \right] \quad (2)$$

where M_S is the saturation magnetization, K_{eff} is the effective anisotropy constant, $f(T_B)$ is the distribution function of T_B and τ_m is the measuring time. In Fig. 3 are shown the simulated curves obtained using eqn (1) and (2).

These results confirm the presence of magnetic interactions among particles³⁵ that displaces T_B to high temperatures. The magnetic interactions such as exchange interaction³⁶ might be negligible due to particle separation, by the presence of PEI on the particle surface. However, the dipolar interparticle interactions,³⁷ which remain important at larger distances, would be the ones that rule the magnetic behavior of the system. In addition, the surface effects are not negligible due to the lack of translational symmetry, the low coordination number of magnetic ions,⁹ and the existence of broken magnetic exchange bonds, which are responsible for the surface spin disorder.³⁸

The irreversibility temperature onset between the experimental ZFC and FC curves is around 295 K that reflects the strong magnetic interaction. The experimental FC curve grows weakly as the temperature decreases and shows a tendency towards saturation below T_B , but also it shows clear differences concerning the theoretical FC curve. This finding corroborates the presence of magnetic interactions^{37,38} in the system.

The hysteresis loops measured at 300 K for PEI-Fe₃O₄ NPs are shown in Fig. 4a. The saturation magnetization (M_S) does not reach a saturation value, neither at 300 K nor 5 K, even at the highest applied field of 7 T. Suggesting that the occurrence of a strong anisotropy field and/or magnetic disorder at the surface layer,³⁹ that makes difficult the alignment along the field direction. To estimate the M_S , we use the approach to saturation model,⁴⁰ providing M_S values of 66.5 Am² kg⁻¹ and 79.4 Am² kg⁻¹ for 300 K and 5 K, respectively. The smaller values in comparison to the one expected for bulk magnetite ($M_S \approx 92$ Am² kg⁻¹)⁴¹ is due to the non-magnetic mass present in our samples. We also observed the absence of coercive field and remanence magnetization above 150 K, confirming the superparamagnetic regime above this temperature, where the thermal energy is predominant and the orientations of magnetic moments are random, so the $M(H)$ curve shows reversible trend.

The $M(H)$ curve at 300 K can be fitted to the Langevin function⁴² since relaxed states are expected. Accordingly, the magnetization is described by $M(H, T) = \int_0^\infty \mu L \left(\frac{\mu H}{kT} \right) f(\mu) d\mu$, where the log-normal distribution of magnetic moments is considered, and the mean magnetic moment ($\langle \mu \rangle$) of each particle is related to its volume (V) by $\langle \mu \rangle = M_S(V)$,^{43–45} where M_S is the saturation magnetization. The best-fitting is achieved with the parameters $\mu_0 = 9747 \mu_B$, and $\sigma = 1.61$ as shown in Fig. 4a, where μ_0 is the median of the distribution related to the mean magnetic moment and σ is the polydispersion index. From these results, we can estimate the mean magnetic

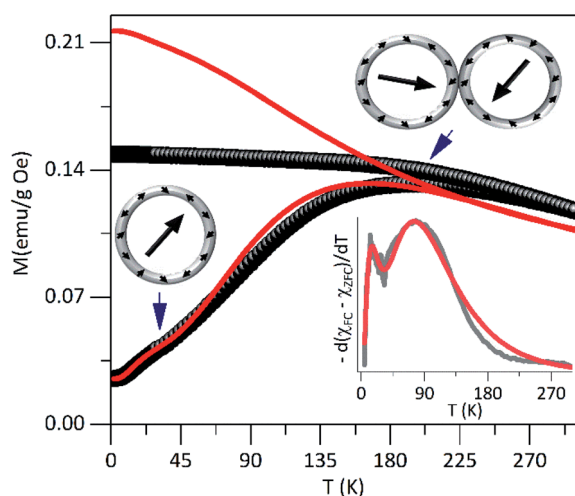


Fig. 3 The dc magnetization curves in ZFC–FC modes for PEI-Fe₃O₄ NPs, carried out at 2.39 kA m⁻¹. In the inset is shown the simulated data fit (red solid lines).



moment value given by $\langle \mu \rangle = \mu_0 \exp(\sigma^2/2)$. Assuming spherical particles and using the equation: $\langle \mu \rangle = M_S \langle V \rangle = \pi M_S \langle D \rangle^3/6$, a value of $\langle D \rangle \approx 10.1$ nm, is estimated for the mean particle size. This value is in excellent agreement with the particle size value determined from TEM data analysis (~ 9.6 nm). Besides, Fig. 4a shows a simulated $M(H)$ curve considering parameters obtained from TEM and the Langevin function. This result suggests that the relaxed states (superparamagnetic state of non-interacting particles) is influenced by the magnetic intra- and interparticle interactions and the magnetic polydispersion index.

In Fig. 4b is showing the H_C vs. T curve obtained from the $M(H)$ curves at different temperatures for the PEI-Fe₃O₄ NPs. As it is observed, the H_C shows an increase with temperature decrease. The inset of Fig. 4b shows the $M(H)$ curve measured at 5 K, with an $H_C \sim 311$ Oe. It is known H_C is very sensitive to factors such as anisotropy type,¹⁴ size and distribution of particles, morphology, surface spin disorder, and interparticle interaction.³⁷ In our sample, we must consider the surface disorder, and the interparticle interaction which delays the thermal relaxation of the magnetic moments of NPs, are predominant factors that determine the value of H_C .

However, the temperature dependence of the coercive field can be modeled taking into account the particle size distribution (distribution of T_B obtained from the ZFC-FC analysis) and/or the interaction effects. The coercive field is given by

$\langle H_C \rangle_T = M_r(T)/(\chi_r + M_r(T)/H_{CB}(T))$, where M_r is the remanence magnetization, χ_r is the susceptibility of the superparamagnetic particles, and H_{CB} is the coercive field of blocked particles. An empirical parameter γ is included in $f(\gamma T_B)$ which is related to the particle interactions.^{34,46} As shown in the main panel of Fig. 4b, the fit provides a $\gamma = 0.8$, which is close to the value expected for systems with negligible interparticle interactions ($\gamma = 1$).⁴⁶

In order to explore possible exchange coupling effects between the disordered surface spins and the magnetic core, we measured the $M(H)$ loops at low temperature after field-cooling the sample with a field $H_{FC} = 1591.5$ kA m⁻¹, to ascertain the presence of an EB anisotropy field (H_{EX}). Fig. 5 shows the presence of an exchange field⁴⁷ for temperatures below ≈ 60 K, with a thermal dependence of exponential type down to $T = 5$ K.

The H_{EX} origin was assigned to the coupling of a layer of disordered spins at the particle surface and a well-ordered region of spin in the core region of the particle. The core spins exert torque on the surface spins that do not follow the anisotropy direction of the core due to the disorder, leading to the H_{EX} occurrence. To evaluate the exchange anisotropy at low temperatures we use the H_{EX} thermal dependence, given by $H_{EX}(T) = H_{EX}(0)\exp(-BT)$ where $H_{EX}(0)$ is the EB field at $T = 0$ K and B is a constant. The fit provides a $H_{EX}(0) = 0.5$ kA m⁻¹ and $B = 5.5 \cdot 10^{-2}$ K⁻¹. The small value of $H_{EX}(0)$ shows the EB effect is rather weak, and it seems to be associated with the magnetic coupling between the disordered surface spins and the magnetic core. This is consistent with previous reports on core-shell Au/Fe₃O₄ NPs where the disordered spins are present at both the inner and outer surface of the magnetite shell in the Au/Fe₃O₄ NPs.³⁴

The H_{EX} low value indicates that the two magnetic regions are weakly coupling, where the effective anisotropy of the core is expected to be greater to achieve the reversal of the surface spins region and where the surface spins rearrange faster than core spins. It is worth mentioning that we are considering just the

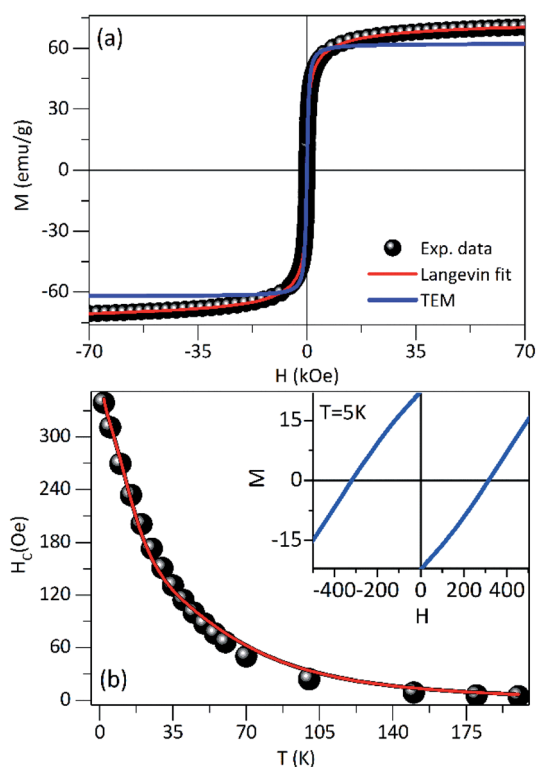


Fig. 4 (a) Magnetization vs. magnetic field loop of the NPs at 300 K. The fit using the Langevin model (red solid line) is included. The modeling with parameters obtained from TEM data analysis is also included (black solid line). (b) The temperature dependence of the coercive field was fitted with the model proposed. The inset shows the hysteresis loops of the NPs obtained at 5 K.

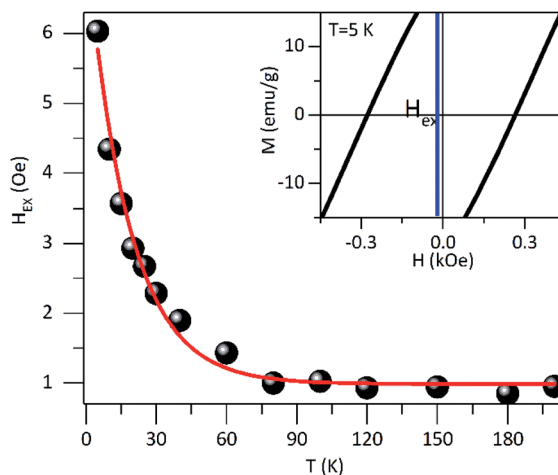


Fig. 5 Exchange bias H_{EX} anisotropy field as a function of the temperature obtained after field cooling the samples with a field of $H_{FC} = 1591.5$ kA m⁻¹. The inset shows the shift of the hysteresis loop at 5 K, which defines the H_{EX} for PEI-Fe₃O₄ NPs.



intrinsic origin of H_{EX} and dipolar interactions do not affect the value of H_{EX} , which could result in H_{EX} reduction in large cooling fields¹⁸ and H_{EX} increase in small particles due to the fact that shell is thicker than core.¹⁷

Fig. 6 shows the in-phase (χ') and the out-of-phase (χ'') susceptibility components as a function of temperature with a range from 0.2 Hz to 1 kHz and in an oscillating magnetic field of 0.08 kA m⁻¹. As was observed, the peak position of $\chi' \times T$ curve is located at ~ 200 K for the lowest frequency (0.2 Hz) and this maximum was shifted to higher temperatures with the frequency increase. This high-temperature peak is expected for NPs with interparticle interactions and this behavior was also observed in $\chi'' \times T$ curve (Fig. 6b). Also, a shoulder in the low temperature that clearly shows a dependence on frequency is evidenced, as shown in the $d\chi''/T \times dT$ curve (inset on Fig. 6b). It could be attributed to non-interacting particles. Other low-

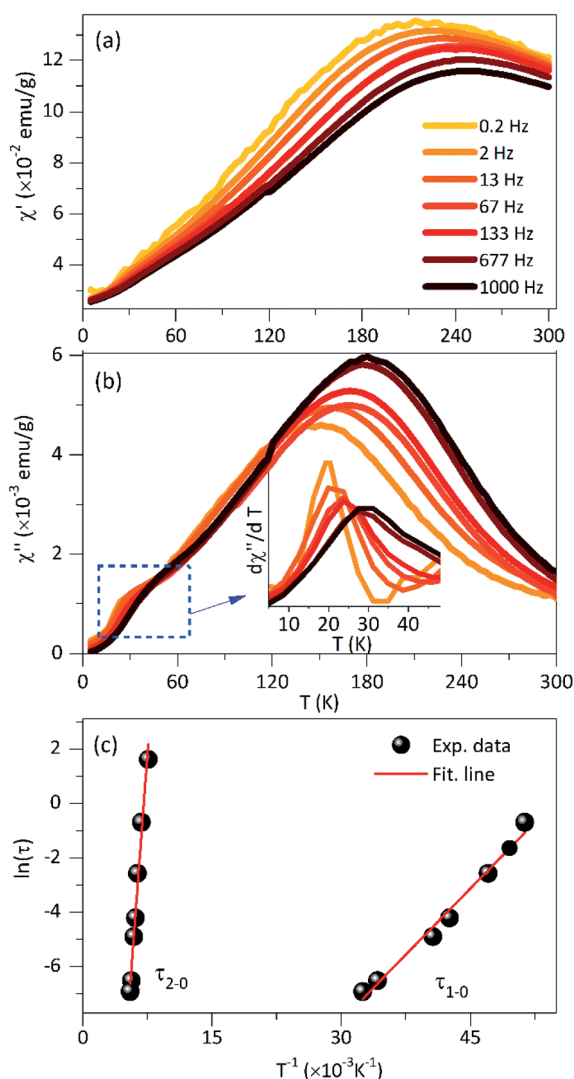


Fig. 6 AC-magnetic susceptibility, (a) in-phase component $\chi'(T)$, at different excitation frequencies, and with an oscillating field of 0.4 kA m⁻¹ for PEI-Fe₃O₄ NPs. (b) The out-of-phase component $\chi''(T)$ curve and the inserted $d\chi''/T \times dT$ curve are shown. (c) Analysis of the relaxation time as a function of the inverse of T_m using the Néel–Arrhenius relation obtained from the imaginary component $\chi''(T)$.

temperature reports were assigned to a significant amount of surface spin disorder at 15 K for 7 nm maghemite NPs⁴⁸ while spin-glass-like-transition at 35 K for 40 nm magnetite NPs.¹⁹

Considering that non-interacting particles where the barrier energy is determined by the uniaxial anisotropy, the spin relaxation is a thermally activated process, as is proposed by the Néel–Arrhenius model, $\tau = \tau_0 \exp(E_a/k_B T)$, where E_a/k_B is the activation energy. Assuming that the temperatures in $\chi''(T, f)$ correspond to the T_B and the low-temperature peak is best observed in $d\chi''/dT$ vs. T curve (inset on see Fig. 6b). The $\ln(\tau) \times 1/T$ curve fit (see Fig. 6c) provides values of $\tau_{1-0} = 1.9 \times 10^{-8}$ s, $E_a/k_B = 325$ K and $\tau_{2-0} = 9.8 \times 10^{-14}$ s, $E_a/k_B = 4224$ K for the low and high temperature peaks, respectively. The characteristic time τ_{1-0} is very close to the values reported for non-interacting NPs in an applied field (10^{-9} – 10^{-11} s) close to zero, while the τ_{2-0} is consistent with interacting NPs.³⁹

The high-temperature peak analysis with the Neél model suggests taking into account the presence of interparticle interactions, and that can be assessed using the Vogel–Fulcher model,⁴⁹ $\tau = \tau_0 \exp(E_a/(k_B(T_{\text{max}} - T_0)))$, where T_0 is a characteristic temperature which magnifies the interaction energy among the NPs and T_{max} is the onset temperature of the blocked state. Considering a characteristic time of $\tau_0 = 10^{-9}$ s, he obtained parameters from the experimental data fit are $T_0 \sim 56$ K and $E_a/k_B \sim 1760$ K. According to what was reported by Yasin *et al.*, when $T_0 > 0$ has a collection of interacting spins; meanwhile, when $T_0 < 0$ indicates a spin-glass system.⁵⁰

Besides that, in the absence of an applied low magnetic field, E_a is given by $E_a = K_{\text{eff}}V$, where K_{eff} is the effective magnetic anisotropy constant. Using the NPs size obtained from TEM data analysis, we estimated a $K_{1-\text{eff}} = 10^4$ J m⁻³ for the low-temperature peak. The $K_{1-\text{eff}}$ value is very close to the magnetocrystalline value of bulk magnetite (1.35×10^4 J m⁻³).⁵¹ In addition, the value obtained from E_a/k_B was used to estimate the NPs mean diameter (considering them as spherical). Using K_{eff} of bulk magnetite, a diameter size of ~ 8.7 nm was calculated

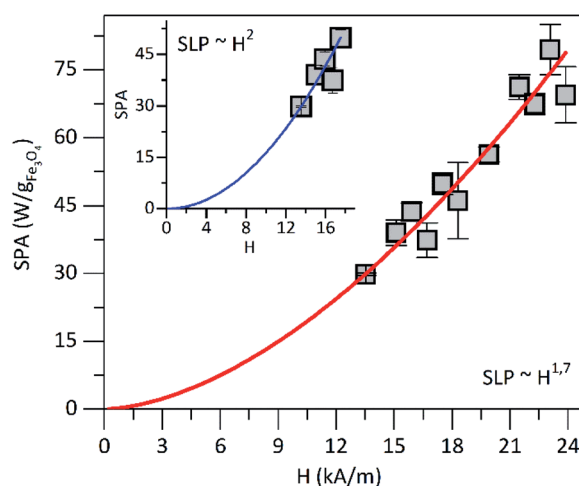


Fig. 7 SLP vs. magnetic field amplitude at 571 kHz. The line presents the best fit using $\text{SLP} \approx H^{1.7}$. In the inset is shown the SLP vs. magnetic field amplitude in the low magnetic field until 17.51 kA m⁻¹.



Table 1 SLP values of magnetic NPs reported in the literature with approximate sizes and different surfaces coatings

Name	Surface coating	Core size [nm]	SLP [W g^{-1}]	Measured conditions [kA m^{-1}]/[kHz]	Ref.
Fe_3O_4	PEI-poly(acrylic acid-co-maleic acid)	10.3	15	7.28/621.7	58
Fe_3O_4	Polyethylene glycol	11.0	4.5	3.2/600	59
Fe_3O_4		10.0	16	\sim 8/524.2	60
Iron oxide ^a	Oleic acid-oleylamine-trioctylphosphine oxide	8.8	80.7	23.87/580.5	61
Fe_3O_4		10.2	63	9.8/276	62
Fe_3O_4	PEI-poly(acrylic acid-co-maleic acid)	10.0	12.15	11.94/111	63
Fe_3O_4	Oleic acid	8.0	33.5	26.6/265	64
Fe_3O_4	PEI	9.6	70	23.87/571	This work

^a The solvent used was hexane and water, in the others.

being this very close to the size found by TEM while \sim 21 nm corroborates the presence of agglomerated NPs.

For further characterization, we used the empirical parameter, ϕ , given by:⁴⁹ $\phi = \Delta T_f / T_f \Delta \log_{10}(f)$, which is the relative shift of the peak temperature (T_f) per AC frequency decade change. The experimental data provides a value of $\phi = 0.14$ for the low-temperature peak. This value is very close to the ones reported for non-interacting particles or superparamagnetic systems ($0.10 < \phi < 0.13$). For the high-temperature peak, a $\phi = 0.071$ was determined, which is in the range of values corresponding to interacting particles ($0.03 < \phi < 0.1$).⁵¹ From these results, we can infer that the studied system contains both non-interacting NPs that present superparamagnetic behavior and interacting NPs associate with the aggregates that show collective response even to the presence of PEI.

3.3 Power absorption

Further characterization was performed to assess the heating ability of PEI- Fe_3O_4 NPs in the aqueous medium at pH 7. In order to determine the magnetic properties impact (as surface effects, interparticle interactions) in the heating efficiency of the superparamagnetic PEI- Fe_3O_4 NPs which is quantified by the specific loss power (SLP). In Fig. 7 is shown the SLP value which increases with the increasing field in agreement with reports in the literature.⁵²

For fields below 17.51 kA m^{-1} and using the power law, the SLP displays a field dependence of $\text{SLP} \sim H^2$ (see the inset in Fig. 7), which is in agreement with the linear response theory (LRT) for superparamagnetic particles as predicted by Roseinsweig.⁵³ However, for all measured field values, the dependence is $\text{SLP} \sim H^{1.7}$, which is not predicted by the LRT. On the other hand, the measured SLP value at 23.87 kA m^{-1} was of $\sim 70 \text{ W g}^{-1}$ for the PEI- Fe_3O_4 NPs studied in this work. In Table 1 is summarized the SLP values reported in the literature for iron oxide NPs with different sizes, surface coatings, AMF, and frequency. The SLP value obtained in this work is comparable to the values reported in the literature as observed in Table 1. Therefore, we concluded that the parameters followed in this work were optimized in such a way that the efficiency to transform electromagnetic energy into heat was higher than those reported in the literature.

Furthermore, the differences among the SLP values seems to be related to the dependence on parameters such as particle size distributions,⁵⁴ coating, NPs aggregation, hydrodynamic size, presence of magnetic interactions,⁵⁵ magnetic anisotropy,⁵⁶ magnetic volume,⁵⁷ and/or different experimental conditions, among others. All these factors can affect the LRT behavior of power loss efficiency.

4. Conclusions

PEI-coated Fe_3O_4 NPs of narrow size distribution and mean size of $\sim 10 \text{ nm}$ were successfully synthesized by a one-step coprecipitation route. The systematic investigation of the magnetic properties suggests the presence of non-interacting particles showing a superparamagnetic behavior with a low blocking temperature ($\sim 35 \text{ K}$) and interacting particles, likely forming agglomerates, with a higher blocking temperature ($> 150 \text{ K}$), in which the surface spin disorder is weak and dominated by interparticle interactions. A high SLP value of $\sim 70 \text{ W g}^{-1}$ was obtained (at 571 kHz and 45 kA m^{-1}). Therefore, the magnetic properties play a crucial role in determining the heating efficiency obtained in this work, which is attractive for future application.

Conflicts of interest

The authors declare no competing financial interest.

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References

- H. Cai, X. An, J. Cui, J. Li, S. Wen, K. Li, M. Shen, L. Zheng, G. Zhang and X. Shi, *ACS Appl. Mater. Interfaces*, 2013, **5**, 1722–1731.
- S. Bucak, D. A. Jones, P. E. Laibinis and T. A. Hatton, *Biotechnol. Prog.*, 2003, **19**, 477–484.
- M. Dowaidar, H. Nasser Abdelhamid, M. Hällbrink, K. Freimann, K. Kurrikoff, X. Zou and Ü. Langel, *Sci. Rep.*, 2017, **7**, 9159.
- Y. Bao, J. A. Sherwood and Z. Sun, *J. Mater. Chem. C*, 2018, **6**, 1280–1290.
- Z. Li, M. Kawashita, N. Araki, M. Mitsumori, M. Hiraoka and M. Doi, *Mater. Sci. Eng., C*, 2010, **30**, 990–996.
- L. León Félix, B. Sanz, V. Sebastián, T. E. Torres, M. H. Sousa, J. A. H. Coaquira, M. R. Ibarra and G. F. Goya, *Sci. Rep.*, 2019, **9**, 4185.
- A. Hervault and N. T. K. Thanh, *Nanoscale*, 2014, **6**, 11553–11573.
- Y. Hadadian, A. P. Ramos and T. Z. Pavan, *Sci. Rep.*, 2019, **9**, 18048.
- R. Kodama, *J. Magn. Magn. Mater.*, 1999, **200**, 359–372.
- G. T. Landi, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**, 014403.
- E. Bonetti, L. Del Bianco, D. Fiorani, D. Rinaldi, R. Caciuffo and A. Hernando, *Phys. Rev. Lett.*, 1999, **83**, 2829–2832.
- S. Bedanta and W. Kleemann, *J. Phys. D: Appl. Phys.*, 2009, **42**, 013001.
- P. Dutta, A. Manivannan, M. S. Seehra, N. Shah and G. P. Huffman, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2004, **70**, 174428.
- B. Aslibeiki, M. H. Ehsani, F. Nasirzadeh and M. A. Mohammadi, *Mater. Res. Express*, 2017, **4**, 075051.
- J. A. De Toro, P. S. Normile, S. S. Lee, D. Salazar, J. L. Cheong, P. Muñoz, J. M. Riveiro, M. Hillenkamp, F. Tournus, A. Tamion and P. Nordblad, *J. Phys. Chem. C*, 2013, **117**, 10213–10219.
- M. Vasilakaki, G. Margaritis, D. Peddis, R. Mathieu, N. Yaacoub, D. Fiorani and K. Trohidou, *Phys. Rev. B*, 2018, **97**, 094413.
- A. Ceylan, C. C. Baker, S. K. Hasanain and S. Ismat Shah, *J. Appl. Phys.*, 2006, **100**, 034301.
- Y. Hwang, S. Angappane, J. Park, K. An, T. Hyeon and J.-G. Park, *Curr. Appl. Phys.*, 2012, **12**, 808–811.
- K. L. López Maldonado, P. de la Presa, E. Flores Tavizón, J. R. Farias Mancilla, J. A. Matutes Aquino, A. Hernando Grande and J. T. Elizalde Galindo, *J. Appl. Phys.*, 2013, **113**, 17E132.
- H. Zhao, C. Zhang, D. Qi, T. Lü and D. Zhang, *J. Dispersion Sci. Technol.*, 2019, **40**, 231–238.
- X. Wang, L. Zhou, Y. Ma, X. Li and H. Gu, *Nano Res.*, 2009, **2**, 365–372.
- A. H. Nguyen, G. N. Abdelrasoul, D. Lin, H. Maadi, J. Tong, G. Chen, R. Wang, A. Anwar, L. Shoute, Q. Fang, Z. Wang and J. Chen, *Appl. Nanosci.*, 2018, **8**, 811–821.
- I. Karimzadeh, M. Aghazadeh, T. Doroudi, M. Ganjali and P. Kolivand, *Curr. Nanosci.*, 2017, **13**, 167–174.
- W. Lu, M. Ling, M. Jia, P. Huang, C. Li and B. Yan, *Mol. Med. Rep.*, 2014, **9**, 1080–1084.
- L. Ma, N. Sun, J. Zhang, C. Tu, X. Cao, D. Duan, A. Diao and S. Man, *Nanoscale*, 2017, **9**, 17699–17703.
- J.-H. Kim, S.-M. Kim, I.-H. Yoon, S.-J. Choi and I. Kim, *Sci. Total Environ.*, 2020, **706**, 136020.
- S. S. Rohiwal, N. Dvorakova, J. Klima, M. Vaskovicova, F. Senigl, M. Slouf, E. Pavlova, P. Stepanek, D. Babuka, H. Benes, Z. Ellederova and K. Stieger, *Sci. Rep.*, 2020, **10**, 4619.
- Z. Chen, Z. Lv, Y. Sun, Z. Chi and G. Qing, *J. Mater. Chem. B*, 2020, **8**, 2951–2973.
- R. Prozorov and T. Prozorov, *J. Magn. Magn. Mater.*, 2004, **281**, 312–317.
- J. Vejpravová, V. Sechovský, J. Plocek, D. Nižňanský, A. Hutlová and J.-L. Rehspringer, *J. Appl. Phys.*, 2005, **97**, 124304.
- G. Muscas, G. Concas, S. Laureti, A. M. Testa, R. Mathieu, J. A. De Toro, C. Cannas, A. Musinu, M. A. Novak, C. Sangregorio, S. S. Lee and D. Peddis, *Phys. Chem. Chem. Phys.*, 2018, **20**, 28634–28643.
- I. J. Bruvera, P. Mendoza Zélis, M. Pilar Calatayud, G. F. Goya and F. H. Sánchez, *J. Appl. Phys.*, 2015, **118**, 184304.
- J. C. Denardin, A. L. Brandl, M. Knobel, P. Panissod, A. B. Pakhomov, H. Liu and X. X. Zhang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **65**, 064422.
- L. León Félix, J. A. H. Coaquira, M. A. R. Martínez, G. F. Goya, J. Mantilla, M. H. Sousa, L. de los S. Valladares, C. H. W. Barnes and P. C. Morais, *Sci. Rep.*, 2017, **7**, 41732.
- B. Mehdaoui, A. Meffre, L.-M. Lacroix, J. Carrey, S. Lachaize, M. Gougeon, M. Respaud and B. Chaudret, *J. Magn. Magn. Mater.*, 2010, **322**, L49–L52.
- K. Nadeem, H. Krenn, T. Traussnig, R. Würschum, D. V. Szabó and I. Letofsky-Papst, *J. Appl. Phys.*, 2012, **111**, 113911.
- S. Pal, S. Chandra, M.-H. Phan, P. Mukherjee and H. Srikanth, *Nanotechnology*, 2009, **20**, 485604.
- A. G. Roca, D. Niznansky, J. Poltierova-Vejpravova, B. Bittova, M. A. González-Fernández, C. J. Serna and M. P. Morales, *J. Appl. Phys.*, 2009, **105**, 114309.
- L. Bordonali, Y. Furukawa, M. Kraken, F. J. Litterst, C. Sangregorio, M. F. Casula and A. Lascialfari, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 174426.
- Y. Melikhov, J. E. Snyder, D. C. Jiles, A. P. Ring, J. A. Paulsen, C. C. H. Lo and K. W. Dennis, *J. Appl. Phys.*, 2006, **99**, 08R102.
- P. Dutta, S. Pal, M. S. Seehra, N. Shah and G. P. Huffman, *J. Appl. Phys.*, 2009, **105**, 07B501.
- G. Mihajlović, K. Aledeat, P. Xiong, S. von Molnár, M. Field and G. J. Sullivan, *Appl. Phys. Lett.*, 2007, **91**, 172518.
- E. F. Ferrari, F. C. S. da Silva and M. Knobel, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1997, **56**, 6086–6093.
- F. C. Fonseca, G. F. Goya, R. F. Jardim, R. Muccillo, N. L. V. Carreño, E. Longo and E. R. Leite, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **66**, 104406.



- 45 J. S. Blázquez, V. Franco, C. F. Conde and A. Conde, *J. Nanosci. Nanotechnol.*, 2007, **7**, 1043–1051.
- 46 W. C. Nunes, W. S. D. Folly, J. P. Sinnecker and M. A. Novak, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2004, **70**, 014419.
- 47 Q.-L. Ye, H. Yoshikawa and K. Awaga, *Materials*, 2010, **3**, 1244–1268.
- 48 T. N. Shendruk, R. D. Desautels, B. W. Southern and J. van Lierop, *Nanotechnology*, 2007, **18**, 455704.
- 49 J. L. Dormann, L. Bessais and D. Fiorani, *J. Phys. C: Solid State Phys.*, 1988, **21**, 2015–2034.
- 50 S. Mohammad Yasin, R. Saha, V. Srinivas, S. Kasiviswanathan and A. K. Nigam, *J. Magn. Magn. Mater.*, 2016, **418**, 158–162.
- 51 G. F. Goya, T. S. Berquó, F. C. Fonseca and M. P. Morales, *J. Appl. Phys.*, 2003, **94**, 3520–3528.
- 52 G. Goya, V. Grazu and M. Ibarra, *Curr. Nanosci.*, 2008, **4**, 1–16.
- 53 R. E. Rosensweig, *J. Magn. Magn. Mater.*, 2002, **252**, 370–374.
- 54 M. Palihawadana-Arachchige, H. Nemala, V. M. Naik and R. Naik, *J. Appl. Phys.*, 2017, **121**, 023901.
- 55 S. Bae, S. W. Lee and Y. Takemura, *Appl. Phys. Lett.*, 2006, **89**, 252503.
- 56 E. Lima, E. De Biasi, R. D. Zysler, M. Vasquez Mansilla, M. L. Mojica-Pisciotti, T. E. Torres, M. P. Calatayud, C. Marquina, M. Ricardo Ibarra and G. F. Goya, *J. Nanopart. Res.*, 2014, **16**, 2791.
- 57 V. M. Khot, A. B. Salunkhe, N. D. Thorat, R. S. Ningthoujam and S. H. Pawar, *Dalton Trans.*, 2013, **42**, 1249–1258.
- 58 I. Y. Tóth, D. Nesztor, L. Novák, E. Illés, M. Szekeres, T. Szabó and E. Tombácz, *J. Magn. Magn. Mater.*, 2017, **427**, 280–288.
- 59 T. Miyazaki, T. Tange, M. Kawashita and B. Jeyadevan, *RSC Adv.*, 2020, **10**, 26374–26380.
- 60 M. Avolio, A. Guerrini, F. Brero, C. Innocenti, C. Sangregorio, M. Cobianchi, M. Mariani, F. Orsini, P. Arosio and A. Lascialfari, *J. Magn. Magn. Mater.*, 2019, **471**, 504–512.
- 61 N. Jović Orsini, B. Babić-Stojić, V. Spasojević, M. P. Calatayud, N. Cvjetičanin and G. F. Goya, *J. Magn. Magn. Mater.*, 2018, **449**, 286–296.
- 62 A. F. Alves, S. G. Mendo, L. P. Ferreira, M. H. Mendonça, P. Ferreira, M. Godinho, M. M. Cruz and M. D. Carvalho, *J. Nanopart. Res.*, 2016, **18**, 27.
- 63 D. Nesztor, K. Bali, I. Y. Tóth, M. Szekeres and E. Tombácz, *J. Magn. Magn. Mater.*, 2015, **380**, 144–149.
- 64 R. Ghosh, L. Pradhan, Y. P. Devi, S. S. Meena, R. Tewari, A. Kumar, S. Sharma, N. S. Gajbhiye, R. K. Vatsa, B. N. Pandey and R. S. Ningthoujam, *J. Mater. Chem.*, 2011, **21**, 13388.

