


Cite this: *RSC Adv.*, 2020, 10, 38351

Microwave-assisted calcination of electrospun indium–gallium–zinc oxide nanofibers for high-performance field-effect transistors†

Seong-Kun Cho  and Won-Ju Cho*

The effects of microwave-assisted calcination of electrospun In–Ga–Zn–O (IGZO) nanofibers intended for electronic devices are unclear. To this end, a systematic study was conducted on the effects of microwave-assisted calcination on the microstructure and optical and mechanical properties of electrospun IGZO nanofibers used in high-performance field-effect transistors (FETs). To clarify the role of microwave annealing (MWA) on the characteristics of the electrospun nanofibers, calcination was carried out using two techniques: MWA and conventional thermal annealing (CTA). The morphological differences between IGZO nanofibers calcined using the two techniques were analyzed by scanning electron microscopy (SEM); the diameter of nanofibers was significantly reduced through MWA as compared to CTA. After calcination, the optical transmittance in the visible region was slightly improved, with the MWA-calcined nanofibers exhibiting a higher transmittance than the CTA-calcined nanofibers. Scratch test results showed that the calcination improved the adhesion strength of the nanofibers to the SiO₂ substrate; MWA was more effective in improving the mechanical properties than CTA. Furthermore, the effects of MWA calcination on the electrical properties of FETs fabricated using the electrospun IGZO nanofibers were investigated. The MWA-calcined devices showed better electrical characteristics and reliability than the CTA-calcined devices for IGZO nanofiber FETs.

Received 5th June 2020
Accepted 17th September 2020

DOI: 10.1039/d0ra04963h

rsc.li/rsc-advances

Introduction

Field-effect transistors (FETs) based on 1D semiconductor nanowires, nanofibers, or nanobelts are extensively studied for future functional devices such as photodetectors, chemical and biological sensors, logic circuits, active matrix organic light emitting diodes, and active channel layers owing to their large surface-to-volume ratio, excellent flexibility, high transparency, and unique crystal structure.^{1–4} Among the numerous 1D nanomaterials, indium–gallium–zinc oxide (IGZO) nanofibers in particular have received significant attention because of their optical transparency over the visible wavelength range, large specific surface area, and excellent electrical performance.^{5–7} Electrospinning is widely used for fabricating IGZO nanofibers because it does not require expensive vacuum equipment, and the processing time is relatively short, thus reducing the cost of the process. Moreover, because it is a simple solution-based process, the composition ratio can be easily controlled.^{8–11} In an electrospinning process, a polymer solution is ejected from a small nozzle under a strong electric field of several kilovolts per centimeter. The accumulation of the static charges on the

surface of the droplets induces a jet, resulting in the formation of continuous nanofibers, which randomly gather on the collector surface.^{12,13} Electrospun IGZO nanofibers incorporated with polymeric organic materials, such as carbon and solvents, exhibit a poor performance as electronic devices, and the weak adhesion due to repeated overlap on the substrate causes the fibers to peel off. Furthermore, the carrier transfer in FETs is confined to a narrow region between the channel and the gate oxide, implying that the interfacial properties of the nanofiber and gate insulator have a significant impact on the performance of IGZO nanofiber FETs.^{14–16} Therefore, calcination using heat treatment is essential for impurity removal, electrical activation, adhesion, and improvement in mechanical properties of electrospun nanofibers.^{17,18} Conventional thermal annealing (CTA) using furnaces has been commonly employed in the calcination of nanofibers. However, a fundamental problem of CTA is that it takes a long time for processing and transfers high heat to the samples. An alternative to this method is microwave annealing (MWA) using electromagnetic waves. In MWA, energy is transferred uniformly to the material, regardless of the conduction medium, enabling fast propagation and energy conversion based on rapid heating processes unlike other thermal annealing processes. Depending on the heating mechanism, microwave heating processes have the following advantages: comparatively low temperature process and volumetric and uniform heating from inner materials.^{19–21} Furthermore, MWA

Department of Electronic Materials Engineering, Kwangjuon University, 20, Gwangun-ro, Nowon-gu, Seoul 01897, Republic of Korea. E-mail: chowj@kw.ac.kr

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ra04963h



is a very efficient technique because it can selectively heat only IGZO without heating by microwave absorption on plastic substrates or glass substrates. Unlike RTA, MWA also has the advantage of being inexpensive because it uses atmospheric-based process equipment. Accordingly, the volumetric heating properties of MWA provide an efficient calcination process for removing impurities, such as the polymeric organic materials in electrospun IGZO nanofibers and are advantageous for improving adhesion to substrates. Few practical studies on the effects of microwave-assisted calcination on electrospun IGZO nanofibers intended for electronic device applications exist.

Therefore, in this study, we investigated the effects of microwave-assisted calcination on the microstructure and optical and mechanical properties of electrospun IGZO nanofibers. In addition, we fabricated bottom gate-type FETs using the electrospun IGZO nanofibers to analyze the effects of MWA calcination on the electrical properties and reliability of the FETs. The results were compared with that of CTA calcination. Finally, the effects of microwave energy on the defects, traps, electron concentration, and reliability of the electrospun IGZO nanofibers were investigated.

Experimental

For the preparation of IGZO nanofiber precursor solutions, indium nitrate hydrate [$\text{In}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$, 99.99%], gallium nitrate hydrate [$\text{Ga}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$, 99.99%], and zinc acetate dehydrate [$\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$, 99.99%] were dissolved in 0.5 ml of *N,N*-dimethylformamide (DMF) in a molar ratio of 2 : 1 : 1, and then stirred at 800 rpm for 1 h at room temperature (RT, 25 °C) using a magnetic stirrer. Ethanol-containing polyvinylpyrrolidone (PVP, MW \approx 1 300 000) was then added and stirred at 800 rpm for 2 h at RT. All the reagents were purchased from Sigma Aldrich and used without further purification. Subsequently, the prepared precursor solution was loaded into a plastic syringe with a pinhead internal diameter of 0.635 mm; the IGZO nanofiber was electrospun onto a p-type Si substrate with a 100 nm-thick thermally grown SiO_2 layer. The experimental setup of the electrospinning system is shown in Fig. 1(a), where the NE-1000 of New Era Pump System is used as the syringe pump. A voltage of 20 kV was applied to the pinhead for electrospinning, and a copper plate placed at distance of 20 cm served as the counter electrode (collector). The syringe pump pressure was set to 0.4 ml h^{-1} , while the temperature and humidity during IGZO nanofiber electrospinning were maintained at 25 °C and 25%, respectively. As a calcination process

to remove solvent residuals and polymer matrix in the electrospun IGZO nanofibers, we performed the MWA of 2.45 GHz at 1800 W for 2 min in an air ambient. Our MW irradiation system is shown in Fig. S1,† and the temperature of the MWA measured by an IR thermometer is shown in Fig. S2.† For comparison, the CTA process was carried out at 600 °C for 30 min in O_2 ambient. Finally, we fabricated IGZO nanofiber FETs by forming source and drain electrodes through a lift-off process after depositing a 150 nm-thick Ti layer with an e-beam evaporator. The schematic of the IGZO nanofiber FETs is shown in Fig. 1(b). The channel width and length of the fabricated FETs are 10 and 20 μm , respectively.

The surface morphology of the IGZO nanofibers was analyzed using scanning electron microscopy (SEM; Sirion 400, FEI Company, Hillsboro, OR). The optical properties of the IGZO nanofibers were investigated using an Agilent 8453 ultraviolet-visible (UV-vis) spectrophotometer in the wavelength range of 300–800 nm. The chemical composition of the IGZO NFs was analyzed by X-ray photoelectron spectroscopy (XPS; PHI 5000 Versa Probe II, ULVAC), performed using monochromatic Al-K α radiation ($\lambda = 0.833$ nm). The crystallinity of the IGZO nanofibers were investigated by X-ray diffraction (XRD; SmartLAB, Rigaku Corporation) at 40 kV and 30 mA using Cu-K α radiation ($\lambda = 0.154$ nm) in the conventional Bragg–Brentano configuration. We evaluated the acoustic emission and friction coefficient to determine the critical load using a scratch test apparatus (CP-4, CETR) with a rockwell-shaped diamond indenter. The electrical characteristics were measured using an Agilent 4156 B Precision Semiconductor Parameter Analyzer. All the measurements were performed in a dark box to suppress external light and electrical noise.

Results & discussion

Fig. 2 shows the temperature profiles of the MWA (solid red line) and CTA (blue dashed line) processes applied to the calcination of IGZO nanofibers. Compared to the CTA method, MWA has a shorter calcination time and a faster ramp up and down rate, resulting in a much lower thermal budget, which is

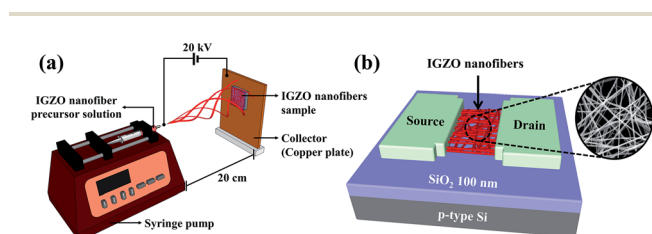


Fig. 1 (a) Experimental set-up of the electrospinning system, and (b) schematic structure of IGZO nanofiber FETs.

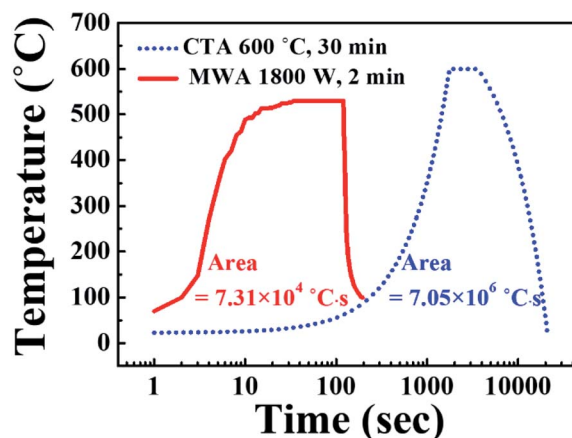


Fig. 2 Temperature profile of MWA and CTA processes for a-IGZO nanofibers calcination.



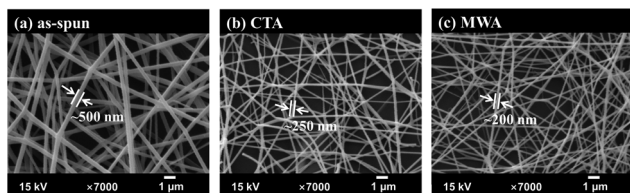


Fig. 3 SEM images of (a) as-spun, (b) CTA calcined and (c) MWA calcined IGZO nanofibers.

expressed as the product of the process time and temperature. In CTA calcination, the IGZO nanofibers are exposed to thermal energy at 600 °C for 30 min and during long ramping up and down steps, resulting in a high thermal budget of 7.05×10^6 °C s. By contrast, in MWA calcination, the IGZO nanofibers are exposed to a thermal energy at 1800 W for 2 min and during short ramping up and down steps. Using an infrared (IR) thermometer, we found the temperature of the sample to be approximately 530 °C under a microwave power of 1800 W, resulting in a low thermal budget of 7.31×10^4 °C s. Therefore, the MWA has a 96.4 times lower thermal budget than the CTA.

Fig. 3 shows the SEM images of the (a) as-spun, (b) CTA-calcined, and (c) MWA-calcined IGZO nanofibers. The diameter of the IGZO nanofibers shown in the SEM image is the average of 20 randomly selected fibers for each condition. The diameter of the IGZO nanofibers in Fig. 3(a) without calcination (as-spun) is approximately 500 nm, whereas the diameters of the CTA and MWA-calcined nanofibers decrease to approximately 250 and 200 nm, respectively, as shown in Fig. 3(b) and (c). In addition, the smaller diameter of the calcined fibers suggests that MWA is more efficient for PVP removal and IGZO calcination, despite the shorter annealing times and lower temperatures than CTA. As a result, we established a nanofiber calcination technique using a low thermal budget and cost-effective MWA process.

Fig. 4 shows the optical properties of the electrospun IGZO nanofibers before and after the calcination. The optical transmittance spectra of the nanofibers in the wavelength range from 300 to 800 nm are shown in Fig. 4(a), where the transmittance of the calcined nanofibers is found to be slightly improved compared to the as-spun fibers. In the visible wavelength range of 400–700 nm, shown in Fig. 4(b), the as-spun and CTA-calcined IGZO nanofibers exhibit similar average transmittances of 89.11 and 89.78%, respectively. The MWA-calcined

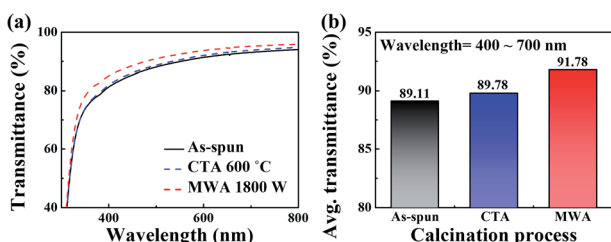


Fig. 4 (a) Optical transmittance spectra of nanofibers and (b) average transmittance in the visible light wavelength range of 400–700 nm.

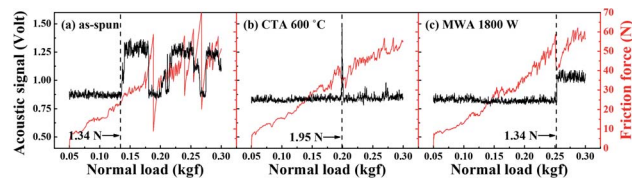


Fig. 5 Scratch test results for (a) as-spun, (b) CTA-calcined and (c) MWA-calcined IGZO nanofibers. The arrow indicates the critical load at the onset of cracking.

nanofibers, on the other hand, exhibit a higher average transmittance of 91.78%.

The effects of calcination on the mechanical properties between the IGZO nanofibers and the SiO₂ substrate were analyzed by conducting a scratch test, as shown in Fig. 5. The scratch test was conducted based on continuous loading with a normal force ranging from 0.5 to 2.9 N. The samples were scratched by increasing the normal load at a rate of 0.08 N s^{−1} with a scratching speed of 0.1 mm s^{−1} for distance of 3 mm. The recorded frictional and acoustic emission signals for the as-spun, CTA, and MWA-calcined fibers are shown in Fig. 5(a), (b) and (c), respectively. The acoustic emission signal consists of an elastic wave generated by a sudden crack in the material when an external load is applied. The fluctuations in the acoustic emission signal indicate the onset of cracking, consistent with the friction data. The irregular pattern observed in the frictional force is probably due to the chipping of the IGZO nanofibers. The critical load of the spun IGZO nanofibers on the SiO₂ substrate was 1.34 N but increased to 1.95 N by CTA calcination and 2.41 N by MWA calcination. The average critical load derived from the scratch test showed that the calcination improved the mechanical strength of the IGZO nanofibers and the adhesion to SiO₂, with MWA calcination being more effective than CTA calcination.

Fig. 6 shows a comparison of the effects of MWA and CTA calcination on the electrical properties of bottom gate-type IGZO nanofiber FETs. The transfer characteristic curves (I_D – V_G) in Fig. 6(a) were measured at a drain voltage (V_D) of 10 V, and the gate voltage (V_G) was swept from −20 to 40 V. In addition, the output characteristic curves (I_D – V_D) in Fig. 6(b) were measured at $V_D = 0$ to 20 V and $V_G - V_{TH} = 0$ to 20 V (at 2 V intervals). The device performance largely depends on the calcination method, without which the device performance is very poor, as shown in Fig. 6(a). Compared to the CTA-calcined FETs, the threshold voltage (V_{TH}) of the MWA-calcined FETs

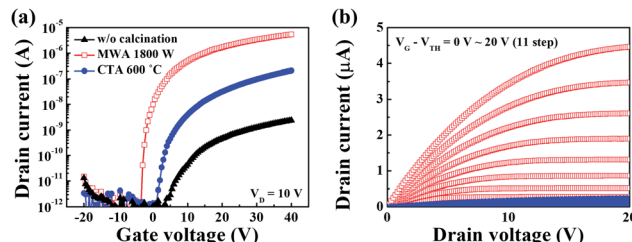


Fig. 6 (a) Transfer and (b) output characteristic curves of CTA and MWA-calcined IGZO nanofiber FETs.



shifts to the left in Fig. 6(a), and the drain current increases significantly, as shown in Fig. 6(a) and (b), indicating that MWA increases the carrier concentration in the IGZO nanofiber channel. The additional electrical properties according to the temperature of the CTA and the power of the MWA are shown in Fig. S3.†

The electrical parameters of the electrospun IGZO nanofiber FETs calcined with MWA and CTA were extracted from the I - V curves and summarized in Table 1. The MWA-calcined IGZO nanofiber FETs showed a field effect mobility (μ_{FE}) of $37.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, a sub-threshold swing (SS) of $304.9 \text{ mV dec}^{-1}$, a V_{TH} of -2.8 V , and an on/off current (I_{ON}/I_{OFF}) ratio of 4.3×10^6 . In comparison, the CTA-calcined IGZO nanofiber FETs exhibited a lower μ_{FE} of $1.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, a higher SS of $986.6 \text{ mV dec}^{-1}$, a higher V_{TH} of 3.1 V , and a lower I_{ON}/I_{OFF} ratio of 1.9×10^5 . In particular, the magnitude of the SS depends on the trap density at the channel/gate dielectric interface (D_{it}) and the total trap density at the bulk channel layer (N_{SS}), which significantly affects the electrical performance and reliability. Thus, we extracted N_{SS} and D_{it} from SS as follows:^{22,23}

$$SS = \frac{qk_B T(N_{SS}t_{ch} + D_{it})}{C_i \log(e)} \quad (1)$$

where q is the electron charge, k_B is the Boltzmann's constant, T is the absolute temperature, and C_i is the gate dielectric capacitance per unit area, which is estimated to be $3.45 \times 10^{-8} \text{ F cm}^{-2}$. To estimate the change in N_{SS} or D_{it} depending on the calcination conditions, N_{SS} or D_{it} is separately set to zero, and thus, the calculated N_{SS} and D_{it} values represent the maximum densities. The MWA and CTA-calcined IGZO nanofiber FETs have D_{it} values of 1.1×10^{12} and $3.6 \times 10^{12} \text{ cm}^{-2} \text{ eV}^{-1}$ and N_{SS} values of 1.6×10^{17} and $5.1 \times 10^{17} \text{ cm}^{-3} \text{ eV}^{-1}$, respectively. Furthermore, we extracted the interface electron trap density (N_{tr}) and electron concentration (N_{ch}) from the nanofiber channel using the turn-on voltage (V_{on}) and the following relationship:^{24,25}

$$N_{tr} = \frac{C_i |V_{TH} - V_{on}|}{q} \quad (2)$$

$$N_{ch} = \frac{C_i |V_{on}|}{qt_{ch}} \quad (3)$$

where t_{ch} is the thickness of the channel layer. As a result, as summarized in Table 1, the extracted N_{tr} values of the calcined

nanofiber channels are 1.5×10^{11} and $5.6 \times 10^{11} \text{ cm}^{-2}$ for MWA and CTA, and the extracted N_{ch} values are 3.1×10^{16} and $5.4 \times 10^{15} \text{ cm}^{-3}$ for MWA and CTA, respectively. Thus, MWA-calcined nanofiber FETs has lower V_{TH} and higher drain drive current (I_D) because N_{tr} is lower and N_{ch} is higher than CTA-calcined nanofiber devices. The electrical properties of the IGZO nanofiber FET, shown in Fig. 6, suggest that the microwave energy effectively reduces the defects and traps and increases the electron concentration in a short processing time, contributing not only to improved performance but also to improved reliability.

Fig. 7 shows the time dependence of the threshold voltage shift (ΔV_{TH}) under positive bias temperature stress (PBTS) and negative bias temperature stress (NBTS) acting on the IGZO nanofiber FETs with respect to the calcination process. Although the long-term instability is critical for electronic device performance, there are few reports on the reliability of electrospun IGZO nanofiber FETs. In the PBTS and NBTS tests, the measurement conditions were $V_{GS} = V_{TH0} \pm 20 \text{ V}$ and $V_{DS} = 0 \text{ V}$ for 10^3 s at 25 – 85°C , where V_{TH0} is the pristine threshold voltage before gate bias stress. In both the MWA and CTA-calcined devices, V_{TH} shifted toward the positive voltage in PBTS and negative voltage in the NBTS with increasing gate bias stress time. In addition, as the bias stress temperature increased, ΔV_{TH} increased faster with the bias stress time. Comparing Fig. 7(a) and (b), we find that the MWA-calcined devices have lower ΔV_{TH} than the CTA-calcined devices. The threshold voltage instability due to gate bias stresses have been reported to be associated with the trapping of electrons or holes in the trap states present at the channel/gate interface or in the gate insulator bulk.^{26,27} In other words, positive ΔV_{TH} by PBTS is due to the oxygen-related acceptor-like trap states, whereas negative ΔV_{TH} by NBTS is due to the oxygen pore-related donor-like trap states. We fitted the experimental data for ΔV_{TH} using the following stretched-exponential equation for charge trapping:²⁸

$$\Delta V_{TH}(t) = \Delta V_{TH0} \left[1 - \exp \left\{ - \left(\frac{t}{\tau} \right)^\beta \right\} \right] \quad (4)$$

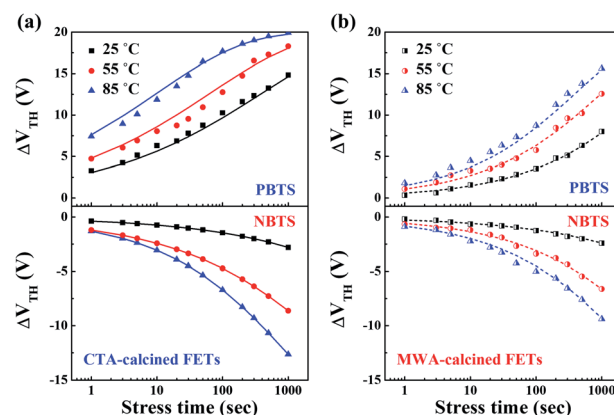


Fig. 7 Time dependence of ΔV_{TH} under PBTS and NBTS tests for the electrospun IGZO nanofiber FETs calcined with (a) MWA and (b) CTA. The solid lines in (a) and the dotted lines in (b) represent the curves fitted with the stretched-exponential equation.

Table 1 Electrical parameters of the electrospun IGZO nanofiber FETs calcined with MWA and CTA and extracted from I - V curves

	MWA	CTA
V_{TH} (V)	-2.8	3.1
N_{ch} (cm^{-3})	3.1×10^{16}	5.4×10^{15}
N_{tr} (cm^{-2})	1.5×10^{11}	5.6×10^{11}
μ_{FE} ($\text{cm}^2 \text{ V}^{-1} \text{ s}^{-1}$)	37.6	1.6
SS (mV dec^{-1})	304.9	986.6
I_{ON}/I_{OFF}	4.3×10^6	1.9×10^5
D_{it} ($\text{cm}^{-2} \text{ eV}^{-1}$)	1.1×10^{12}	3.6×10^{12}
N_{SS} ($\text{cm}^{-3} \text{ eV}^{-1}$)	1.6×10^{17}	5.1×10^{17}



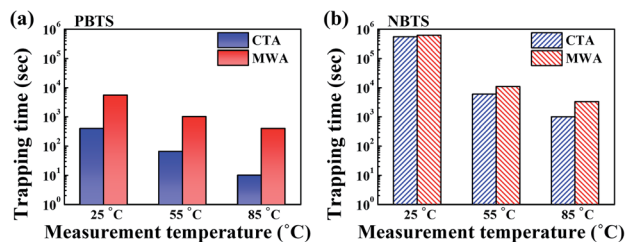


Fig. 8 Charge trapping time (τ) of IGZO nanofiber FETs calcined with MWA and CTA under (a) PBTS and (b) NBTS tests.

where ΔV_{TH0} is the $\Delta V_{TH}(t)$ at infinite time, τ is the characteristic trapping time of the carriers, β is the stretched-exponential exponent, and t is the stress time. The solid lines in (a) and the dashed lines in (b) fitted with the stretched-exponential equations are in good agreement with the experimental results, indicating that τ depends on the temperature.

In the stretched-exponential equation, the characteristic trapping time (τ) of a thermally activated carrier is expressed as:

$$\tau = \tau_0 \exp\left(\frac{E_\tau}{k_B T}\right) = \nu^{-1} \exp\left(\frac{E_\tau}{k_B T}\right) \quad (5)$$

where the thermal activation energy is given by $E_a = E_\tau \beta$; E_τ is the average effective energy barrier that needs to be overcome before the electrons in the nanofiber channel enter the gate insulator; τ_0 is the thermal prefactor; and ν is the frequency prefactor for emission across the barrier. The τ value of the IGZO nanofiber FETs calcined with MWA and CTA under (a) PBTS and (b) NBTS tests is shown in Fig. 8. As the test temperature increases, τ gradually decreases, and the MWA-calcined IGZO nanofiber FETs have a higher τ value than the CTA-calcined devices. This indicates that the CTA-calcined IGZO nanofiber FETs trap the carrier faster than the MWA-calcined devices, resulting in long-term reliability issues.

Fig. 9 shows an Arrhenius plot of $\ln(\tau)$ versus the inverse of the absolute temperature in (a) PBTS and (b) NBTS tests for IGZO nanofiber FETs calcined with MWA and CTA. The linear relationship between $\ln(\tau)$ and $1/T$ represents a thermally activated charge trapping process in the IGZO nanofiber FETs, and the slope of the Arrhenius plot represents E_τ for charge transport under the PBTS and NBTS tests. The E_τ value of the MWA-calcined nanofiber FETs is 0.40 eV for PBTS and 0.72 eV for NBTS. The E_τ value of the CTA-calcined nanofiber FETs is

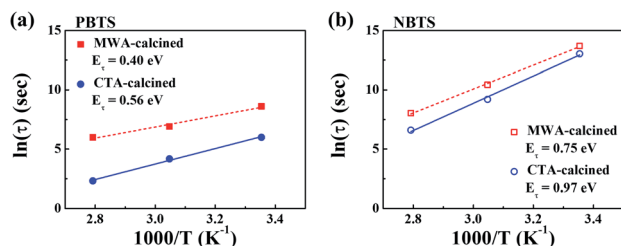


Fig. 9 Arrhenius plots of $\ln(\tau)$ versus the inverse of the absolute temperature in (a) PBTS and (b) NBTS tests for IGZO nanofiber FETs calcined with MWA and CTA.

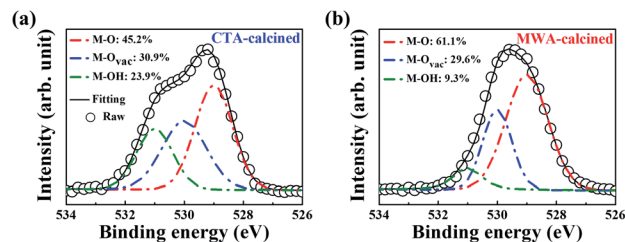


Fig. 10 XPS O 1s spectra of (a) CTA-calcined and (b) MWA-calcined IGZO nanofibers.

0.56 eV for PBTS and 0.97 eV for NBTS. The magnitude of E_τ is related to the lattice arrangement of the IGZO nanofiber channels. Lower E_τ indicates fewer defects in the channel and a more optimal lattice arrangement, resulting in improved instability.^{29,30} Therefore, we conclude that MWA calcination is more effective in improving the stability of electrospun IGZO nanofiber devices than CTA calcination.

Fig. 10 exhibits the XPS O 1s spectra of the IGZO nanofibers treated by (a) CTA calcination and (b) MWA calcination. XPS analysis was performed to determine the chemical composition of the nanofibers, and the spectra were collected after etching the surface of IGZO nanofibers to a depth of several nanometers with Ar^+ ions, which is to reduce the errors due to surface contamination. The O 1s peak was deconvoluted into three separate chemical components, taking into account the composition of the precursor solution of electrospun IGZO NF. The main peak at 529 eV represents stoichiometric oxygen (M–O), the subpeak at 530 eV represents the oxygen vacancies (M–O_{vac}), and the subpeak at 531 eV is associated with loosely bound oxygen impurities (M–OH), such as chemically adsorbed oxygen, H_2O , and CO_3 .³⁰ As a result, the proportions of M–O_{vac} and M–OH were higher and that of M–O much smaller in the case of the MWA-calcined nanofibers as compared with those for the CTA-calcined nanofibers. It is known that the oxygen deficiency associated with M–O_{vac} binding decreases the stability during the PBTS and NBTS tests, while loosely coupled oxygen impurities associated with M–OH binding act as charge traps and reduce the on-current of FETs.^{31,32} Therefore, we believe that the MWA calcination process

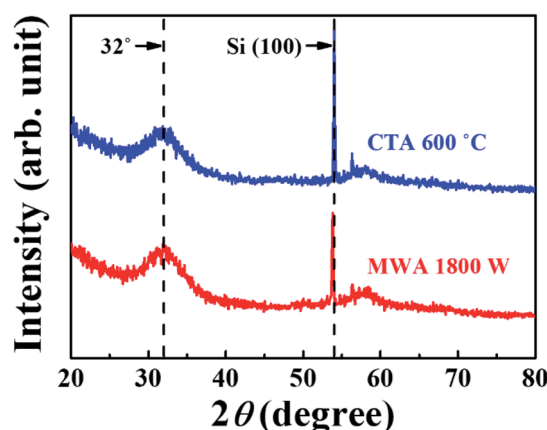


Fig. 11 XRD patterns of CTA-calcined and MWA-calcined IGZO nanofibers.



is very effective in improving the electrical properties and stability of electrospun IGZO nanofiber FETs.

Fig. 11 illustrates the XRD patterns of IGZO nanofibers for the calcination method. Except for the sharp peak due to the substrate (100) Si single crystal at 55° , no peaks due to the IGZO crystal were found in both the CTA and MWA calcination processes. Instead, only an amorphous-like IGZO broad peak appeared at 32° . These results suggest that IGZO nanofibers exist in an amorphous state regardless of the calcination process. Thus, this result implies that impurities dominate the device performance, not crystallinity or grain size. In addition, the calcination is an important process for determining device performance because it has an absolute influence on these impurities.

Conclusions

We fabricated IGZO nanofiber FETs using electrospinning and introduced an MWA technique for calcination. The MWA allowed rapid calcination of the electrospun IGZO nanofibers and dramatically reduced the thermal budget delivered to the device. The diameter of the MWA-calcined IGZO nanofibers was found to be 200 nm, which is lower than that of the CTA-calcined IGZO nanofibers, suggesting that MWA is more effective for PVP removal and IGZO nanofiber calcination. In particular, we improved the mechanical performance with excellent adhesion and the optical characteristics with high transmittance through MWA calcination. Furthermore, MWA calcination significantly improved the electrical properties, such as V_{TH} , N_{ch} , μ_{FE} , SS, I_{ON}/I_{OFF} , D_{it} , and N_{SS} , and enhanced the reliability under PBTS and NBTS compared with the CTA process. In conclusion, the low thermal-budget MWA technique with high energy transfer efficiency can be actively applied to IGZO nanofibers as an effective calcination method.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Research Foundation of Korea (NRF) grant funded by the Korea government (MSIT) (No. 2020R1A2C1007586). This research was funded and conducted under the Competency Development Program for Industry Specialists of the Korean Ministry of Trade, Industry and Energy (MOTIE), operated by Korea Institute for Advancement of Technology (KIAT) (No. P0002397, HRD program for Industrial Convergence of Wearable Smart Devices).

Notes and references

- 1 Z. Y. Yuan and B. L. Su, *Colloids Surf., A*, 2004, **241**, 173.
- 2 H. Hou and D. H. Reneker, *Adv. Mater.*, 2004, **16**, 69.
- 3 E. P. S. Tan and C. T. Lim, *Compos. Sci. Technol.*, 2006, **66**, 1102.

- 4 Y. Li, X. Y. Yang, Y. Feng, Z. Y. Yuan and B. L. Su, *Crit. Rev. Solid State Mater. Sci.*, 2012, **37**, 1.
- 5 D. Li and Y. Xia, *Nano Lett.*, 2003, **3**, 555.
- 6 H. Jiang, J. Ma and C. Li, *Adv. Mater.*, 2012, **24**, 4197.
- 7 N. G. Cho and I. D. Kim, *Sens. Actuators, B*, 2011, **160**, 499.
- 8 D. Li, Y. Wang and Y. Xia, *Nano Lett.*, 2003, **3**, 1167.
- 9 S. S. Mali, H. Kim, W. Y. Jang, H. S. Park, P. S. Patil and C. K. Hong, *ACS Sustainable Chem. Eng.*, 2013, **1**, 1207.
- 10 A. Frenot and I. S. Chronakis, *Curr. Opin. Colloid Interface Sci.*, 2003, **8**, 64.
- 11 N. Bhattarai, D. Edmondson, O. Veisoh, F. A. Matsen and M. Zhang, *Biomaterials*, 2005, **26**, 6176.
- 12 H. Wu, L. Hu, M. W. Rowell, D. Kong, J. J. Cha, J. R. McDonough, J. Zhu, Y. Yang, M. D. McGehee and Y. Cui, *Nano Lett.*, 2010, **10**, 4242.
- 13 N. Bhardwaj and S. C. Kundu, *Biotechnol. Adv.*, 2010, **28**, 325.
- 14 Y. Meng, G. Liu, A. Liu, Z. Guo, W. Sun and F. Shan, *ACS Appl. Mater. Interfaces*, 2017, **9**, 10805.
- 15 F. Wang, L. Song, H. Zhang, Y. Meng, L. Luo, Y. Xi, L. Liu, N. Han, Z. Yang, J. Tang and F. Shan, *Adv. Electron. Mater.*, 2018, **4**, 1700336.
- 16 A. Liu, Y. Meng, H. Zhu, Y. Y. Noh, G. Liu and F. Shan, *ACS Appl. Mater. Interfaces*, 2017, **10**, 25841.
- 17 V. Thavasi, G. Singh and S. Ramakrishna, *Energy Environ. Sci.*, 2008, **1**, 205.
- 18 N. A. Barakat, B. Kim and H. Y. Kim, *J. Phys. Chem. C*, 2008, **113**, 531.
- 19 C. J. Ko, Y. K. Lin and F. C. Chen, *Adv. Mater.*, 2007, **19**, 3520.
- 20 L. F. Teng, P. T. Liu, Y. J. Lo and Y. J. Lee, *Appl. Phys. Lett.*, 2012, **101**, 132901.
- 21 J. Y. Pyo and W. J. Cho, *J. Korean Phys. Soc.*, 2018, **73**, 978.
- 22 L. Shao, K. Nomura, T. Kamiya and H. Hosono, *Electrochem. Solid-State Lett.*, 2011, **14**, H197.
- 23 C. S. Fuh, P. T. Liu, W. H. Huang and S. M. Sze, *IEEE Electron Device Lett.*, 2014, **35**, 1103.
- 24 Y. H. Lin, H. Faber, S. Rossbauer and T. D. Anthopoulos, *Appl. Phys. Lett.*, 2013, **102**, 193516.
- 25 D. A. Koutsouras, N. A. Hastas, D. H. Tassis, C. A. Dimitriadis, P. Frigeri, S. Franchi, E. Gombia and R. Mosca, *J. Appl. Phys.*, 2005, **97**, 064506.
- 26 W. T. Chen, S. Y. Lo, S. C. Kao, H. W. Zan, C. C. Tsai, J. H. Lin, C. H. Fang and C. C. Lee, *IEEE Electron Device Lett.*, 2011, **32**, 1552.
- 27 K. Ghaffarzadeh, A. Nathan, J. Robertson, S. Kim, S. Jeon, C. Kim, U. I. Chung and J. H. Lee, *Appl. Phys. Lett.*, 2010, **97**, 113504.
- 28 C. Avis and J. Jang, *Electrochem. Solid-State Lett.*, 2011, **14**, J9.
- 29 J. K. Jeong, H. Won Yang, J. H. Jeong, Y. G. Mo and H. D. Kim, *Appl. Phys. Lett.*, 2008, **93**, 123508.
- 30 K. W. Jo and W. J. Cho, *Appl. Phys. Lett.*, 2014, **105**, 213505.
- 31 W. G. Kim, Y. J. Tak, B. Du Ahn, T. S. Jung, K. B. Chung and H. J. Kim, *Sci. Rep.*, 2016, **6**, 23039.
- 32 H. Cheong, S. Ogura, H. Ushijima, M. Yoshida, N. Fukuda and S. Uemura, *AIP Adv.*, 2015, **5**, 067127.

