


Cite this: *RSC Adv.*, 2020, 10, 31271

Water-based synthesis of photocrosslinked hyaluronic acid/polyvinyl alcohol membranes via electrospinning†

Karine Cappuccio de Castro,^a Jonny Burga-Sánchez,^b Maria Gabriela Nogueira Campos^c and Lucia Helena Innocentini Mei^a

Electrospinning is a versatile and low-cost technique widely used in the manufacture of nanofibrous polymeric membranes applied in different areas, especially in bioengineering. Hyaluronic acid (HA) is a biocompatible natural polymer, but it has rheological characteristics that make the electrospinning process difficult. Thus, its association with another polymer such as poly(vinyl alcohol) (PVA) is an alternative, as PVA has good rheological properties for electrospinning. Based on this, the aim of this work was to produce, by the conventional electrospinning method, cross-linked HA/PVA membranes free from organic solvent with a low degradation rate in PBS 7.4 solution after the photocrosslinking process and without using any organic solvent. The results showed that the electrospinning occurred effectively for all conditions tested, but the best result for complete cross-linking only occurred with 15 and 30% crosslinker, which was evidenced by infrared spectroscopy. The addition of crosslinker favored the stability of the electrospinning jet, especially for 30% crosslinker concentration. The membranes did not show cytotoxicity even after the cross-linking process, which indicates that the material has potential as a drug delivery device.

Received 4th June 2020
Accepted 3rd August 2020

DOI: 10.1039/d0ra04950f

rsc.li/rsc-advances

1. Introduction

Recently, electrospinning has attracted great attention as it is an effective technique to produce nanofibrous membranes for a large variety of applications, especially in the pharmaceutical and medical fields,¹ including wound dressings,² drug delivery systems,³ and tissue engineering scaffolds.⁴

The study of the physical parameters that involve the electrospinning process, such as jet formation, as a function of electrostatic field strength; fluid viscosity; solvent composition, and molecular weight of polymers in solution, is of great importance as they directly impact the characteristics of the formed nanofibers.^{5,6} In addition to physical properties, polymers suitable for electrospinning should also present the desired characteristics for the final application. Specifically, for wound dressings that are utilized predominantly to improve the cicatrization environment in its various stages; the wound

surface must be covered by dressing materials to enhance and speed up the healing process.^{2,7}

Hyaluronic acid (HA) is a natural polysaccharide, commonly found in connective tissues in the body such as vitreous, umbilical cord, joint fluid, presenting very high viscosity and high surface tension in solution, even at very low concentrations.^{2,8} It consists of repeating disaccharide units of D-glucuronic acid and N-acetyl-D-glucosamine. Due to its unique rheological properties, biodegradability, and biocompatibility, HA has been the subject of study for many biomedical applications including drug administration, dermatological applications, wound dressings, and cell growth scaffolds.^{9–11}

Poly(vinyl alcohol) (PVA) is a synthetic polymer and has received great attention from several areas due to its low cost, high hydrophilicity, and excellent chemical resistance. PVA has been widely used for several biological applications due to its advantages such as non-toxic, non-carcinogenic, biodegradable, and bio-adhesive characteristics, as well as easy processing. As a result of the aforementioned features, PVA can simulate natural tissues and thus be accepted in body implants.^{3,12}

The production of nanofibers with HA in its native form by the conventional method of electrospinning is not viable due to the rheological properties of the HA solution. Pure HA nanofibers were reported by Um *et al.* (2004) and by Wang *et al.* (2005) by a blowing-assisted electrospinning process and the maximum reported HA concentration was 2.7%, even after adapting the process.^{11,13} In this work, we were able to

^aFaculty of Chemical Engineering, Department of Materials Engineering and Bioprocess, University of Campinas, Cidade Universitária Zeferino Vaz, Campinas, SP, Brazil. E-mail: cappuccio.karine@gmail.com

^bPiracicaba Dental School, Physiological Science Department, University of Campinas, Piracicaba, SP, Brazil

^cNanoScience Technology Center, University of Central Florida, Orlando, USA

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ra04950f



electrospinning high molecular weight HA at 2% by conventional electrospinning pathways when associated with PVA, which improves the rheological behavior of HA. In addition, PVA also improved the mechanical properties of HA membranes without altering their good compatibility for biological uses.

Electrospinning of the HA/PVA blend has already been reported with the use of surfactants or solvent mixtures.^{14,15} However, to our knowledge, this is the first study where no organic solvent or surfactant was used to prepare HA and PVA polymeric electrospinnable solutions (only water was used as a solvent), which reduces the toxicity of the final material.

The use of hydrophilic polymers is extremely advisable for making biomedical devices due to their solubility in water as well as the low-toxicity of the system. On the other hand, a highly soluble system will leach very quickly, making it impossible to use effectively when there is a need for contact with body fluids. Therefore the cross-linking process can be an interesting alternative to improve the device's stability and mechanical properties at physiological conditions. As our ultimate goal is to apply the herein prepared membranes as drug carriers for sustained release, we performed a photo-crosslinking step using maleic anhydride as a cross-linking agent.

Given this context, this study demonstrated that it is possible to obtain HA/PVA membranes by the conventional electrospinning technique, using high concentration of high molecular weight HA, and only water as solvent. The electrospinning process was stable and enabled obtaining dressings with desirable properties for applications in the biomedical field as drug carriers. This result has not been consolidated in the Literature yet.

2. Materials and methods

2.1. Materials

Photoinitiator 2-hydroxy-4'-(2-hydroxyethoxy)-2-methylpropiophenone (Irgacure 2959, 98%, $M_w = 224.25 \text{ g mol}^{-1}$ Sigma Aldrich), polyvinyl alcohol (PVA, $M_w \approx 130 \text{ kg mol}^{-1}$, 86.5–89.5 mol% hydrolysis, Vetec), hyaluronic acid sodium salt (HA, cosmetic grade, Symbios, $M_w > 1.6 \text{ MDa}$, D-glucuronic acid $\geq 45\%$, bacterial), maleic anhydride (MA, $M_w \approx 98 \text{ g mol}^{-1}$, 98%, Dinâmica), saline phosphate buffer solution (PBS, pH 7.4 ± 0.2 , Dinâmica), 3-(4,5-dimethylthiazol-2-yl)-2,5-diphenyltetrazolium bromide (Thermo Fisher Scientific Inc., Oregon, USA), calcein AM (component A) (Thermo Fisher Scientific Inc., Oregon, USA), ethidium homodimer-1 (component B) (Thermo Fisher Scientific Inc., Oregon, USA), were used without further purification.

2.2. Solution preparation

PVA (6% w/v) was dissolved in distilled water and stirred on a hot plate shaker at 80°C for 2 h. HA solution (2% w/v) was prepared in NaCl aqueous solution (1% w/v), and at room temperature. PVA and HA were mixed in the proportion of 75PVA : 25HA, under vigorous agitation and at room temperature. Both MA (5%, 15%, 30% by weight of PVA and HA) and Irgacure 2959 (5% by weight of PVA and HA) were added to the

final solution and mixed under vigorous stirring overnight and at room temperature before the electrospinning process. The proportion of 75PVA/25HA was chosen after preliminary studies once it showed greater stability and yield in the electrospinning process.

2.3. Characterization of solutions

The electrical conductivity of the solutions was determined by an Analion K0392 conductivity meter at 25°C . Viscosity was measured in a Haake RheoStress 1 rheometer at 25°C at a speed of 2 rpm to 5 rpm (keeping the torque at 50%). A measurement sensor system having a cylindrical rotor and pot with diameters of 3.6 mm and 4.8 mm respectively was utilized for PVA/HA membranes without crosslinker (spindle SC4-2) and pot with diameters of 3.2 mm and 9.4 mm respectively was utilized for PVA/HA membranes with a crosslinker (spindle SC4-34). Surface tension was measured by the ring method in a K6 Kruss Tensiometer.

2.4. Electrospinning step

The prepared electrospinning solutions were loaded in a 10 mL glass syringe with a metal needle (0.80 mm diameter inner). The distance between the needle tip and the aluminum-foil wrapped plane collector ($10 \times 30 \text{ cm}$) was adjusted to 15 cm (solution with and without a cross-linking agent). A high voltage current of 25 kV (DC) was applied between the needle and the collector to initiate the charged polymer solution jet, which was rate-controlled by the syringe pump. This value was fixed for all solutions, but with the addition of the cross-linking agent, it was possible to have a stable process and without changes in morphology with a voltage of 20 kV. The feed rate was fixed at 0.3 mL h^{-1} and 1 mL h^{-1} for solutions without a cross-linking agent and with a cross-linking agent, respectively. All experiments were performed at room temperature and relative humidity of 20 to 30%. The prepared nanofibers were carefully collected and stored at room temperature until the final cross-linking step.

2.5. Photocrosslinking reaction

In order to activate the photocrosslinking reaction in the fibers, all membranes were placed in a UV light reactor (BLACK LIGHT 8W F8T5BL – 300 mm at 365 nm – PHILIPS) for 40 min and then stored in a desiccator at room temperature for further analysis.

2.6. Characterization of electrospun hyaluronic acid/poly(vinyl alcohol) membranes

2.6.1. Scanning electron microscopy (SEM). The surface morphology of electrospun HA/PVA membranes was investigated by SEM (model Leo 440i – LEO Electron Microscopy/Oxford). The membranes were coated with 200 \AA of gold (Sputter Coater EMITECH, model K450). To determine the diameter distribution, the ImageJ® software was used. The statistical analyzes were made in two SEM photomicrographs for each case.



2.6.2. Swelling behavior. The dried electrospun membranes were initially weighed and immersed into PBS 7.4 for 72 hours at room temperature to attain an equilibrium swelling state. Before measuring the weight of the swollen samples, the excess amounts of waters onto the surface were carefully removed by blotting with filter paper. The swelling ratio was calculated as follows:

$$\% \text{ swelling} = \frac{w_t - w_o}{w_o} \times 100$$

where w_t and w_o are the weights of swollen and dried samples, respectively.

2.6.3. Fourier transform infrared spectroscopy (FTIR). Infrared spectra were collected on a spectrometer (Thermo Scientific – Nicolet 6700 – Madison/USA) and recorded from 500 to 4000 cm^{-1} . The occurrence of the esterification reaction (cross-linking) and inter- and intramolecular interactions between the polymers were characterized by the presence of characteristics functional groups of HA and PVA in the membranes.

2.6.4. Cell viability assay. The culture was grown according to the methodology described by Zhao.¹⁶ Once in confluence, immortalized human keratinocytes (HaCaT) were transferred to 24-well cell culture plates, at a concentration of 5×10^4 cells per mL and incubated at 37 °C in 5% CO_2 for 24 h. After the incubation period, the HA/PVA membranes, with different concentrations of cross-linking agent, were suspended inside the wells and remained in the culture medium in contact with the cells for 24 h. Briefly, the wells were washed with PBS (pH 7.4), and then 500 μL of DMEM with 0.5 mg mL^{-1} of MTT sodium salt was added. The plates were incubated for 3 h in MTT reduction. The medium was removed, the cells were washed with PBS again and 200 μL of absolute ethanol was added to each well to dissolve the formazan crystals. Optical density was measured using a microplate reader wavelength of 570 nm.

2.6.5. Live/dead assay. Live and dead cells were stained with calcein-AM (2 μM) and ethidium homodimer-1 (4 μM), respectively. Briefly, cells seeded on membranes (5×10^4 cells per membrane) and cultured for 24 hours were washed with PBS (pH 7.4) and incubated in live/dead solution for 30 minutes in an incubator at room temperature, protected from light. Excess dye was washed off using PBS and the cells were observed under inverted fluorescent microscope Zeiss Axiovert 40 CFL coupled to an AxioCam MEC camera (Carl Zeiss, Germany).

3. Results and discussion

Before defining the PVA/HA best ratio for the electrospinning process, a preliminary study was performed. Three proportions of PVA/HA were studied: 25/75, 50/50, and 75/25. In all proportions, the electrospinning process occurred and nanofibers were formed, but higher proportions of hyaluronic acid left the solution highly viscous, which caused instability in the electrospinning jet and gave rise to beaded nanofibers (Fig. 1). So, the condition chosen for this study was the one that

presented the formation of uniform nanofibers and stability in the electrospinning process (75/25).

Table 1 shows the rheological parameters of the cross-linking solutions with maleic anhydride (MA). These parameters play an important role in the properties of the formed nanofibers. Therefore, we carefully discussed their effects on the morphology of the nanofibrous membranes obtained in this work.

The addition of MA to the polymeric solution improved the electrospinning parameters, decreased the applied voltage, which was initially 25 kV and went to 20 kV, and the solution ejection flow, which went from 0.3 mL h^{-1} to 1 mL h^{-1} , since it reduces the solutions' viscosity and surface tension, facilitating its flow by the formation of a more homogeneous and defect-free system. MA addition also increased the solution electrical conductivity, causing it to have sufficient surface loads for Taylor cone generation, and consequently the formation of a stable jet. However, it was observed that above 30% of cross-linker the solution was no longer electrospinnable, resulting in a phenomenon similar to electrospray, *i.e.*, the deposition of small droplets in the collector.

SEM photomicrographs showed that increasing MA concentration decreased nanofiber diameter (Table 2) and made diameter distribution more homogeneous (Fig. 2). Although the average diameters are not statistically different, it is possible to observe a tendency of fiber diameter reduction for high crosslinker concentration (Table 2). In addition, it was possible to observe a greater interconnection of nanofibers with the increase of crosslinker concentration, suggesting that the membrane's porosity also decreased. However, additional studies on pores size distribution are under investigation.

Séon-Lutz *et al.* (2019) found similar fiber diameters (348 ± 51 nm) in their study with the addition of EDC and NHS as PVA-HA-HP β CD membrane's crosslinker, however with a modified electrospinning system.² In another work developed by Abdel-Mohsen *et al.* (2019) the fiber diameters were relatively smaller for the PVA/hyaluronan-AgNPs (225 nm) system. In this case, two points must be taken into consideration; first, the electrospinning system was modified and the voltage used was 58 kV; second, hyaluronan served only as reducing and stabilizing agent for the formation of silver nanoparticles, which made its concentration low in the composition.¹⁷

In our work, the membranes were submerged in a PBS 7.4 solution to verify their integrity (Fig. 3a), and as observed the 0 and 5% MA membranes dissolved instantly, while the 15 and 30% remained undissolved after 72 hours (Fig. 3b). Fig. 3c shows the swelling behavior of 15 and 30% MA membranes. Interestingly, in the first 12 hours, 15% MA membrane increased 5 times its initial weight ($S\% = 501$), while 30% MA swelled 296%. However, after 24 hours, 30% MA membrane reached around 550% swelling percentage and maintained it until 72 hours. On the other hand, 15% MA started showing a decrease in the swelling rate, and swelled 381% at 72 hours, suggesting weight loss. Therefore, the degree of cross-linking plays an important role in the membranes' water uptake kinetics and stability.



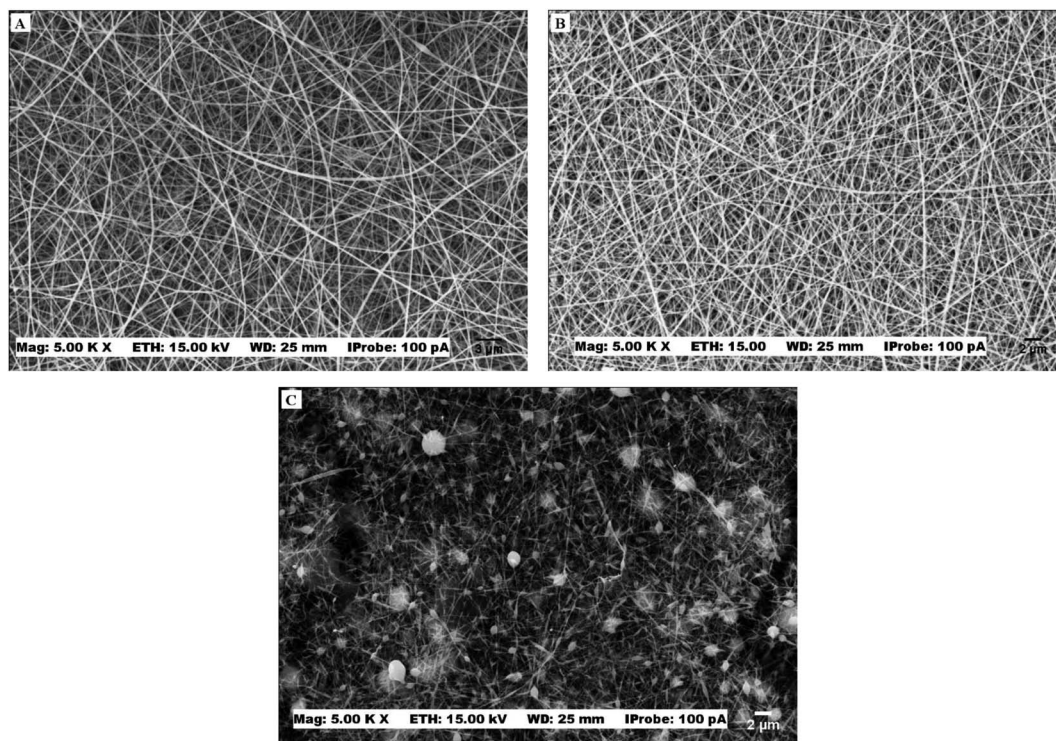


Fig. 1 SEM images of PVA/HA membranes with different proportions (A) 75PVA/25HA, (B) 50PVA/50HA (C) 25PVA/75HA.

This is an important result for the application of membranes to prolonged healing wounds. As observed in the SEM micrographs of Fig. 4, the fibers maintained their integrity; however, there were some points of coalescence and fibers swelling, especially after 72 hours. Séon-Lutz *et al.* (2019) found similar results after 48 h, but in their work, the cross-linking was activated by temperature in the presence of EDC and NHS.²

FTIR analysis (Fig. 5) was important to confirm the cross-linking reaction between PVA and HA chains and the effects of crosslinker concentration, by following the functional groups and the bonds present in the sample.

The spectrum revealed the main bands related to PVA and HA, as expected, since they are the major components of the membranes. The broadband around $3200\text{--}3400\text{ cm}^{-1}$ may be related to the lengthening of the --OH bond and also to the lengthening of the --NH , as the cross-linking process increases with the addition of higher concentrations of MA, the intensity of this band decreases.^{2,18} The weak band close to 2900 cm^{-1} attributed to

the stretching of the --CH vibration, also changes during the cross-linking process. Characteristic bands between 1600 and 1700 cm^{-1} are related to bond stretching of amide carbonyl (--C=O) and carboxylic acid (--C=O) groups of the HA.^{19,20} Some characteristic bands near 1434 and 1089 cm^{-1} for HA are due to amide-type --N--H bending and --C--O--C-- bending vibrations, respectively² and the peak near 1420 cm^{-1} correspond to the stretching of carboxyl bonds.^{21,22} All spectra show PVA characteristic peaks near 2945 cm^{-1} , 1734 cm^{-1} , and 1250 cm^{-1} , relative to alkyl, acetyl, and --C--O groups, respectively.^{23,24} The cross-linking was evidenced by the peak at 1700 cm^{-1} , which is a clear indication of the existence of the --CO--O-- (Fig. 5 – Region I) stretching, most evident in both PVA/HA samples with MA concentrations of 15 and 30%.^{18,25,26} Another proof of cross-linking was the change in intensity at the peak of 1091 cm^{-1} , which is also relative to the cross-linking between --OH of the PVA structure and --C=O from MA molecule, giving rise to the --C--O--C-- band (Fig. 5 – Region II).^{26,27} Another interesting observation was the disappearance of the peak at 1640 cm^{-1} , related to the --C=C-- bond in MA molecule, suggesting that it was also involved in the photocrosslinking reaction.²⁸

Table 1 Rheological parameters of the polymeric solutions of PVA/HA with cross-linking agent

Samples	Superficial tension (mN m^{-1})	Electric conductivity (mS cm^{-1})	Viscosity (cP)
0% MA	45.3 ± 0.6	4.8 ± 0.3	4263 ± 18
5% MA	41.8 ± 0.8	6.0 ± 0.1	212 ± 1
15% MA	39.3 ± 0.6	10.8 ± 0.2	167 ± 2
30% MA	32.3 ± 0.6	15.3 ± 0.2	124.5 ± 0.3

Table 2 Average diameter of nanofibers before and after addition of different MA concentrations

	0% MA	5% MA	15% MA	30% MA
Diameter (nm)	308 ± 74	247 ± 101	226 ± 83	217 ± 65
Minimum diameter (nm)	162	110	108	100
Maximum diameter (nm)	595	576	499	448



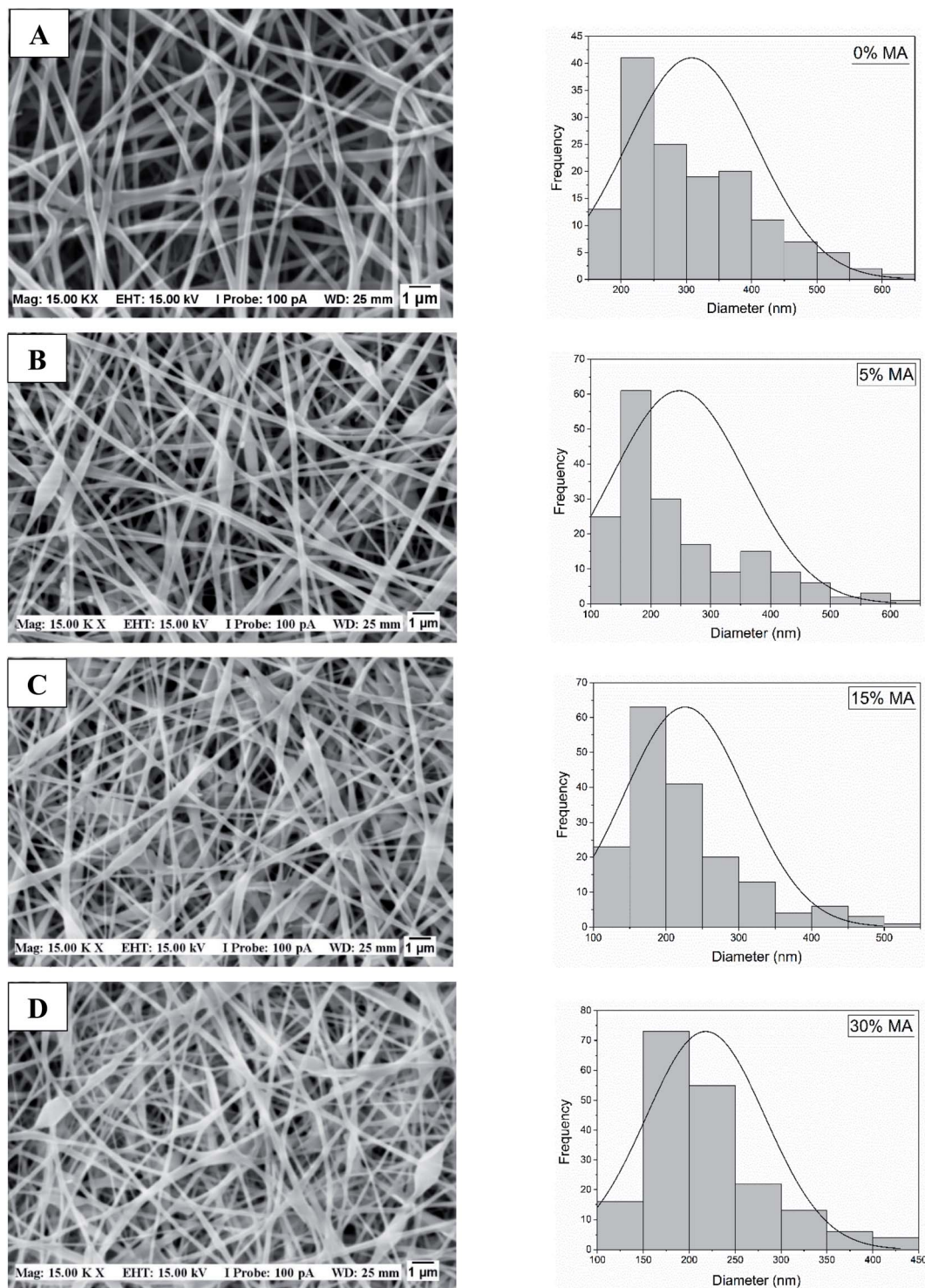


Fig. 2 SEM images of PVA/HA membranes with different MA concentrations (A) 0%, (B) 5% (C) 15% and (D) 30%. The relative humidity during the process varied between 20 and 30% for all cases.

These two types of reactions are the possible cross-linking routes that can occur in the studied systems (Fig. 6). To better understand these processes, the PVA/MA, HA/MA, and PVA/HA casting membranes were made separately. The casting membranes were prepared using the same concentrations used in the solutions that were electrospun. Membrane PVA/MA:

5 mL of PVA solution (6% w/v) were mixed by magnetic stirring with MA (30% relative to the mass of PVA) for 1 hour and poured into a Petri dish. Membrane HA/MA: 5 mL of HA solution (2% w/v) were mixed by magnetic stirring with MA (30% relative to the mass of HA) for 1 hour and poured into a Petri dish. Membrane PVA/HA: PVA (6% w/v) and HA (2% w/v) were

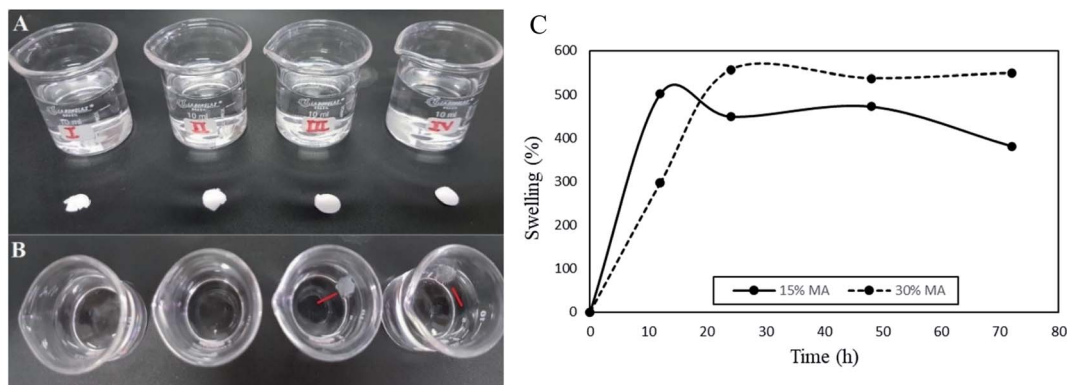


Fig. 3 Illustrative photos of membranes before (A) and after (B) immersion in PBS 7.4 solution for 72 hours. From left to right: (I) 0% MA, (II) 5% MA, (III) 15% MA and (IV) 30% MA. (C) Swelling behavior of 15% MA and 30% MA membranes at 12, 24, 48 and 72 hours.

mixed by magnetic stirring at a ratio of 75 : 25 for 1 hour and poured into a Petri dish. For all formulations, the process was performed under ambient conditions, without the addition of heat or UV light. It was verified that at room temperature the esterification reaction does not occur because the membranes were dissolved in PBS in less than 30 minutes. The same assay was performed with UV radiation and 5% of Irgacure 2959 was added concerning the mass of the polymer in solution. The solutions were then poured into the Petri dish and subjected to UV light for 40 minutes. As a result, the esterification reaction was efficient only for PVA.

Other aspects that may have contributed to the cross-linking stage were presented by Yang *et al.* (2008), in their studies, they showed that PVA and MA were efficiently cross-linked during the electrospinning process, without the need of heat treatment or UV radiation. PVA and MA reacted to form mono-esters or bis-esters during electrospinning, where the high electric field can stimulate the chemical activity of the molecules and cause the rapid evaporation of water, which can accelerate the esterification reaction and, as a consequence, increase the equilibrium constant.²⁹ It is noteworthy that the esterification reaction between (ii) PVA and MA is more likely to occur than the esterification reaction between (i) HA and MA or (iii) PVA and

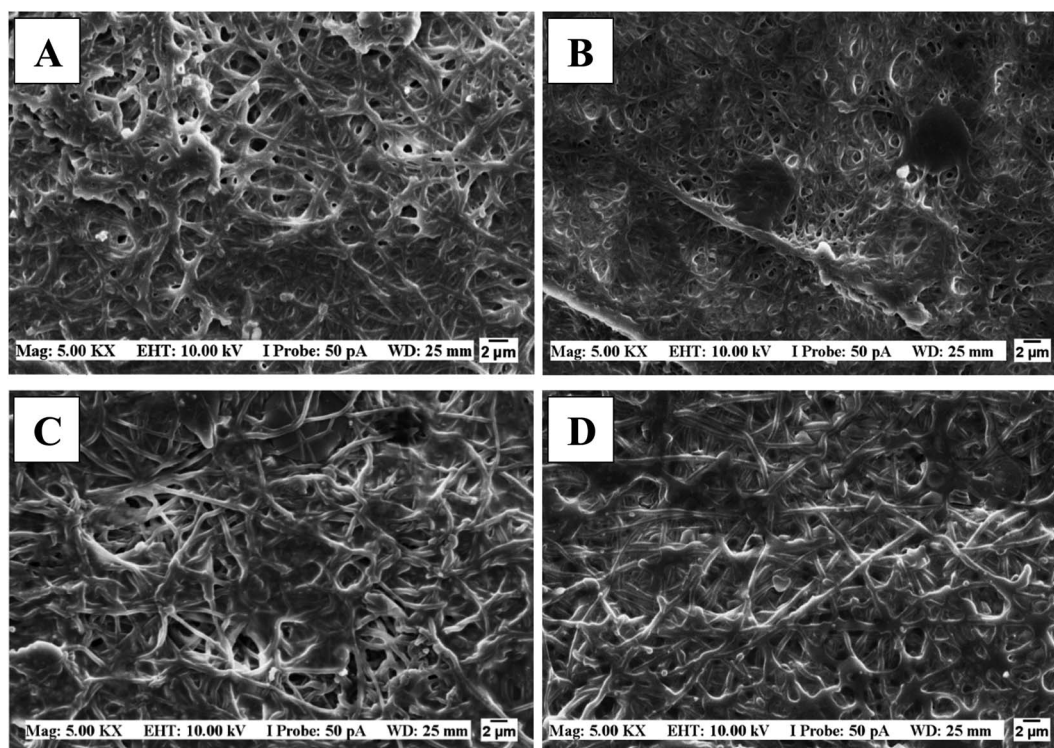


Fig. 4 SEM images of PVA/HA membranes: after 48 hours of immersion (A) 15% MA, (B) 30% MA. After 72 hours of immersion (C) 15% MA and (D) 30% MA.



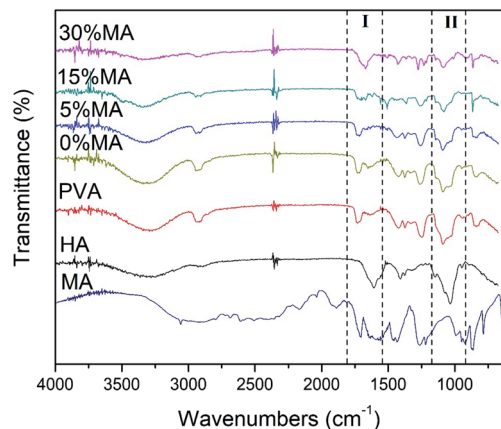


Fig. 5 FTIR spectra of the nanofibrous membranes and the respective formulation components.

HA since the reaction conditions are less favorable to (i) and (iii) than for (ii) (Fig. 6).

The hyaluronic acid cross-linking reaction usually occurs only after HA modification or if the conditions of the medium are favorable, such as the addition of an acidic catalyst together with an activating agent, or in the presence of acid anhydrides in basic medium,^{30,31} which is not the case of the system studied in this paper.

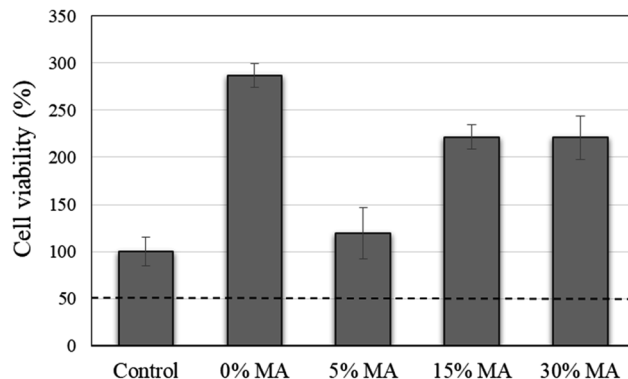


Fig. 7 Evaluation of cell viability of HaCaT cells after being cultured with the electrospun membranes for a period of 24 h.

Although the cross-linking reaction by Fig. 6(ii) occurs in the presence of the electric field, photocrosslinking was carried out to ensure a more effective cross-linking, since we have PVA and HA functional groups competing for the possible active reaction sites.

The HA/PVA membranes did not show direct cytotoxicity against HaCaT cells (Fig. 7). Indeed, they stimulated cell growth in comparison to the negative control of cytotoxicity (no treatment). Comparing the prepared membranes, the one without the cross-linking agent (MA 0%) showed the best cell viability

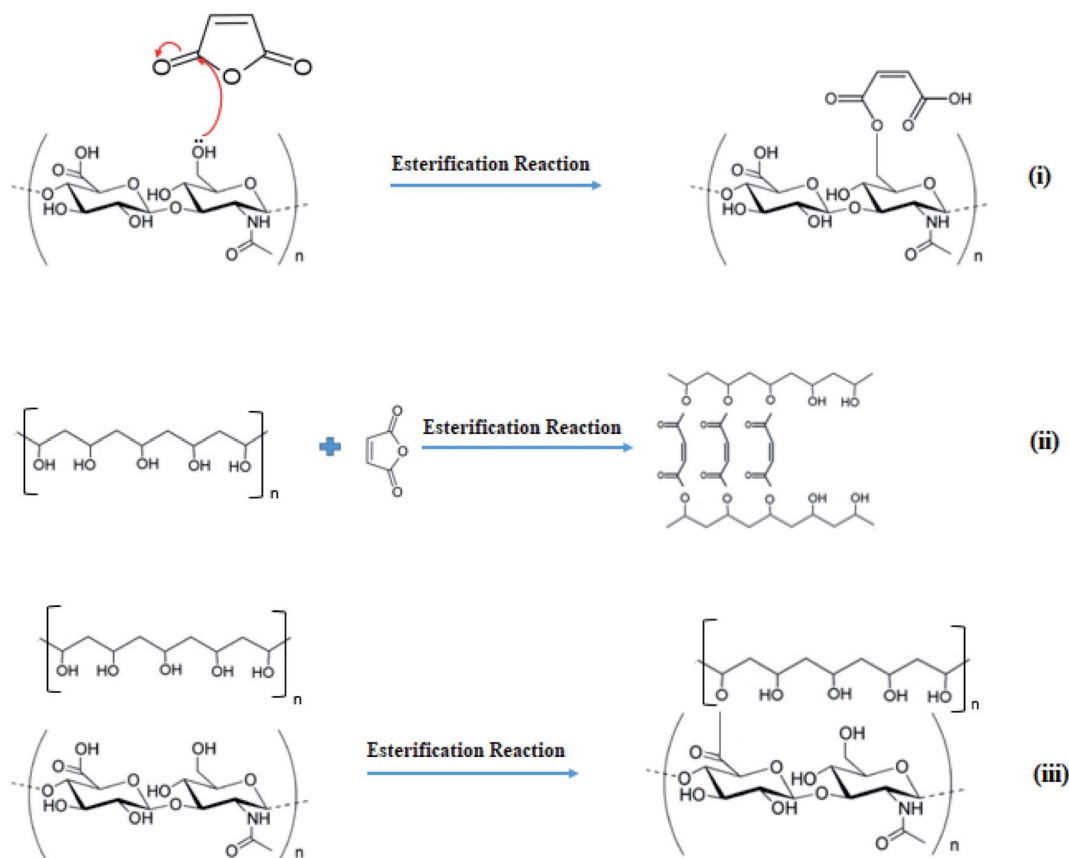


Fig. 6 Schematic representation of possible esterification reactions between (i) HA and MA, (ii) PVA and MA, (iii) PVA and HA.

result (290%), while membranes with MA concentrations of 15 and 30% showed similar cell viability ($\sim 225\%$). The membrane with 5% crosslinker showed less cell viability (120%), which was already expected since the cross-linking reaction was incomplete in this formulation (Fig. 3), as already discussed in the solubility test and FTIR analysis. It is known that unbound chemicals, such as crosslinker agents can be toxic to cells or prevent cell growth.

Live/dead assay was performed to analyze the cell survival at the surface of the membranes, by simultaneously staining live (green-labeled) and dead (red-labeled) cells. The images (Fig. 8) show that cells remain viable when in contact with membranes. Using the Fiji supplement from ImageJ® software, it was possible to calculate the percentage of live and dead cells from

Table 3 Percentage of live cells using image processing software Fiji (ImageJ® extension)

	0% MA	5% MA	15% MA	30% MA
Live cells (%)	99.5	97.1	92.4	83.9
Dead cells (%)	0.5	2.9	7.6	16.1

each sample (Table 3). These results corroborate the data obtained in the MTT assay, thus confirming the biocompatibility of these membranes. As the degree of cross-linking increases, the number of viable cells decreases, this may be related to the fact that the cross-linked membrane decreases water retention and HA, which is the protein present in the skin's ECM and with

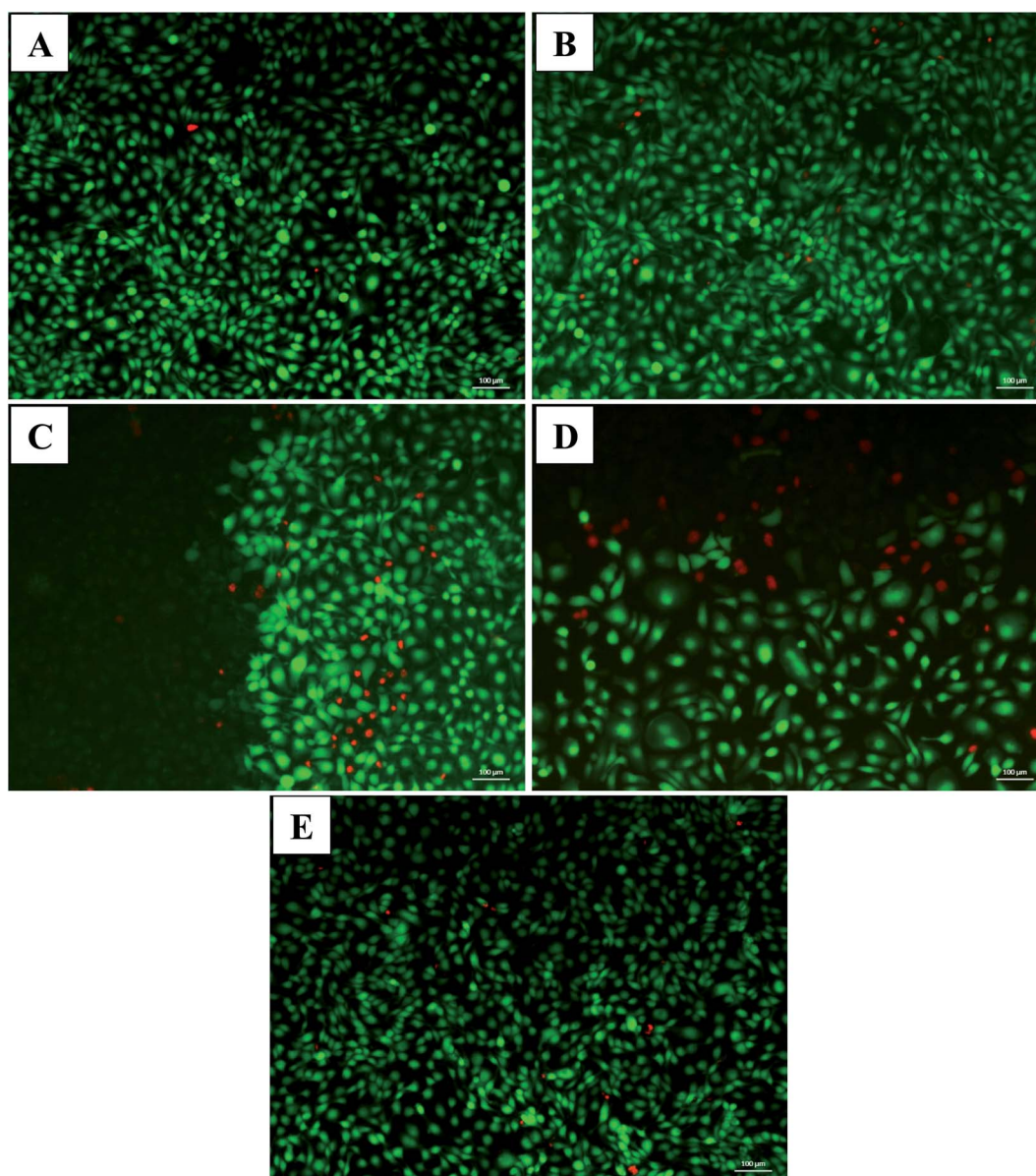


Fig. 8 Fluorescence microscopy images from a live/dead assay of HaCaT cultured on the surface of the produced HA/PVA membranes with different MA concentrations after 24 hours. (A) 0%, (B) 5% (C) 15%, (D) 30% and (E) control. Green channel: viable cells; red channel: dead cells.



excellent biological properties, not be free to stimulate migration and cell adhesion.¹⁴

The fact that the membranes have been successfully cross-linked and have demonstrated no cytotoxicity against human keratinocytes is an excellent indication for this material to be used as a potential carrier for sustained drug release.

4. Conclusion

The electrospinning of PVA/HA membranes was successful and the addition of a cross-linking agent caused an effective improvement in the rheological properties of the solution and improved the process conditions. This was extremely important especially in increasing the ejection rate of the polymer solution from 0.3 mL L⁻¹ to 1.0 mL h⁻¹, which is related to energy-saving and membrane production time. It was possible to obtain homogeneous nanofibers without the presence of beads, using only water as a solvent. Complete cross-linking reaction occurred with the addition of 15 and 30% crosslinker, meaning that the range of crosslinker concentration between 15 and 30% is interesting to investigate, focusing the application on controlled release modulated devices with different duration times before degrading. In addition, the membranes showed high cell viability, indicating that their use *in vivo* may be promising.

Conflict of interest

The authors declare that they have no conflict of interest.

Acknowledgements

The authors thank CNPq (140567/2019-4) and CAPES for financial support. They also thank the Faculty of Chemical Engineering at the University of Campinas.

References

- 1 L. J. Villarreal-Gómez, J. M. Cornejo-Bravo, R. Vera-Graziano and D. Grande, *J. Biomater. Sci., Polym. Ed.*, 2016, **27**, 157–176.
- 2 M. Séon-Lutz, A.-C. Couffin, S. Vignoud, G. Schlatter and A. Hébraud, *Carbohydr. Polym.*, 2019, **207**, 276–287.
- 3 J. Jalvandi, M. White, Y. Gao, Y. B. Truong, R. Padhye and I. L. Kyratzis, *Mater. Sci. Eng., C*, 2017, **73**, 440–446.
- 4 B. Jankovic, J. Pelipenko, M. Skarabot, I. Musevic and J. Kristl, *Int. J. Pharm.*, 2013, **455**, 338–347.
- 5 S. Babitha, L. Rachita, K. Karthikeyan, E. Shoba, I. Janani, B. Poornima and K. Purna Sai, *Int. J. Pharm.*, 2017, **523**, 52–90.
- 6 S. Thakkar and M. Misra, *Eur. J. Pharm. Sci.*, 2017, **107**, 148–167.
- 7 N. Bhardwaj and S. C. Kundu, *Biotechnol. Adv.*, 2010, **28**, 325–347.
- 8 J. Li, A. He, C. C. Han, D. Fang, B. S. Hsiao and B. Chu, *Macromol. Rapid Commun.*, 2006, **27**, 114–120.
- 9 C. Feng, C. Liu, S. Liu, Z. Wang, K. Yu and X. Zeng, *Tissue Eng., Part A*, 2019, **25**, 1289–1299.
- 10 G. D. Prestwich, *J. Controlled Release*, 2011, **155**, 193–199.
- 11 I. C. Um, D. Fang, B. S. Hsiao, A. Okamoto and B. Chu, *Biomacromolecules*, 2004, **5**, 1428–1436.
- 12 J. A. Quinn, Y. Yang, A. N. Buffington, F. N. Romero and M. D. Green, *Polymer*, 2018, **134**, 275–281.
- 13 X. Wang, I. C. Um, D. Fang, A. Okamoto, B. S. Hsiao and B. Chu, *Polymer*, 2005, **46**, 4853–4867.
- 14 S. P. Miguel, D. Simões, A. F. Moreira, R. S. Sequeira and I. J. Correia, *Int. J. Biol. Macromol.*, 2019, **121**, 524–535.
- 15 L. Vítková, L. Musilová, E. Achbergerová, A. Minařík, P. Smolka, E. Wrzecionko and A. Mráček, *Polymers*, 2019, **11**, 1517.
- 16 F. Li, Y. Zhao and Y. Song, in *Nanofibers*, InTech, 2010, p. 438.
- 17 A. M. Abdel-Mohsen, D. Pavliňák, M. Čileková, P. Lepcio, R. M. Abdel-Rahman and J. Jančář, *Int. J. Biol. Macromol.*, 2019, **139**, 730–739.
- 18 B. Zeytuncu, M. Morcali, S. Akman and O. Yucel, *J. Serb. Chem. Soc.*, 2015, **80**, 97–106.
- 19 E. Larrañeta, M. Henry, N. J. Irwin, J. Trotter, A. A. Perminova and R. F. Donnelly, *Carbohydr. Polym.*, 2018, **181**, 1194–1205.
- 20 Y. Zhao, Z. Fan, M. Shen and X. Shi, *Adv. Mater. Interfaces*, 2015, **2**, 1–9.
- 21 R. Gilli, M. Kacuráková, M. Mathlouthi, L. Navarini and S. Paoletti, *Carbohydr. Res.*, 1994, **263**, 315–326.
- 22 S. A. de Oliveira, B. C. da Silva, I. C. Riegel-Vidotti, A. Urbano, P. C. de Sousa Faria-Tischer and C. A. Tischer, *Int. J. Biol. Macromol.*, 2017, **97**, 642–653.
- 23 S. K. Rai and P. Basak, *Int. Conf. Syst. Med. Biol. ICSMB 2010 - Proc.*, 2010, pp. 360–364.
- 24 J. M. Gohil, A. Bhattacharya and P. Ray, *J. Polym. Res.*, 2006, **13**, 161–169.
- 25 Y. B. Truong, J. Choi, J. Mardel, Y. Gao, S. Maisch, M. Musameh and I. L. Kyratzis, *Macromol. Mater. Eng.*, 2017, **302**, 1–9.
- 26 R. F. Majidi, N. Nezafati and M. Pazouki, in *6th International Conference on Nanotechnology (ICN2017)*, Dubai, 2017, pp. 1–4.
- 27 M. Pingrui, C. Cuixian, Y. Lixin, L. Jiding and J. Weijun, *Tsinghua Sci. Technol.*, 2000, **5**, 172–175.
- 28 F. J. Liou and Y. J. Wang, *J. Appl. Polym. Sci.*, 1996, **59**, 1395–1403.
- 29 E. Yang, X. Qin and S. Wang, *Mater. Lett.*, 2008, **62**, 3555–3557.
- 30 R. Buffa, V. Ľ Velebn, L. Pospisilova, E. Prikopova, M. Pravda, P. Nikodym and L. Palek, WO 2010/105582A1, 2010, pp. 1–18.
- 31 A. V. Sochilina, A. G. Savelyev, P. A. Demina, N. V. Ierusalimsky, D. A. Khochenkov, R. A. Akasov, N. V. Sholina, E. V. Khaydukov and A. N. Generalova, *J. Phys.: Conf. Ser.*, 2018, **1124**, 1–6.

