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Strain forces tuned the electronic and optical properties in GaTe/MoS₂ van der Waals heterostructures†

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Two-dimensional (2D) van der Waals heterostructures (vdWHs) have attracted widespread attention in fundamental materials science and device physics. In this work, we report a novel GaTe/MoS₂ vdWH and theoretically investigate the electronic and optical properties based on first-principles calculations. GaTe/MoS₂ vdWH possesses an indirect band gap with type-II band alignment. Meanwhile, the interfacial charge transfer from MoS₂ to GaTe can effectively separate electrons and holes. Also, this vdWH shows improved visible-ultraviolet optical absorption properties compared with those of the isolated GaTe or MoS₂ monolayers. More remarkably, the biaxial strain can not only modulate the band gap but also enhance the optical performance in GaTe/MoS₂ vdWH. In particular, the tensile strain is more effective for improving the optical absorption in the visible light region. These findings indicate that GaTe/MoS₂ vdWH is a promising candidate for nanoelectronics and optoelectronic devices.

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1. Introduction

Since graphene was firstly synthesized experimentally,¹ two-dimensional (2D) layered materials have received widespread attention in nanoelectronics and optoelectronic devices for their excellent physical properties.^{2–4} So far, a wide variety of 2D materials have been extensively studied theoretically and experimentally, such as silicene, stanene, phosphorene, MXenes, transition metal dichalcogenides (TMDCs), and so on.^{5–7} Inspired by the successful isolation of various 2D materials, the van der Waals heterostructures (vdWHs) composed of two 2D materials have attracted intensive research interest, especially for vdWHs composed of two TMDC monolayers.^{8–10} Various vdWHs with different electronic structures have been thoroughly studied, and different novel 2D application devices, such as vertical transistors, solar cells, resonant tunneling diodes, and photodetectors, have been reported.^{11–13} It is well known that the function of the vdWHs deepens largely on their band structures. The separation of electrons and holes into different layers can avoid electron–hole recombination if the vdWHs have type-II band arrangement, which is very useful for the photocatalysis of unipolar electronic devices.^{13,14} Thus, it is necessary to explore new vdWHs to satisfy the multiple device design demands.

Recently, a novel heterostructure composed of group IIIA monochalcogenides and TMDCs have shown great potential in optoelectronic applications. For example, Li *et al.* reported that a heterostructure composed of GaSe and MoS₂ can effectively transport and separate photo-generated charge carriers between layers, resulting in an excellent photovoltaic response.¹⁵ Zhang *et al.* found that the band alignment of the InSe/MoS₂ vdWHs can be controlled by applying external strain for photocatalysis.¹⁶ Meanwhile, Lei *et al.* revealed another novel WTe₂/HfS₂ vdWHs, in which the electronic features can be modulated by applying external strain and electric field.¹⁷ All the above results indicate that vdWHs composed of 2D monochalcogenides and TMDCs monolayers may have great potentials in practical applications.

TMDCs are promising building blocks for the new vdWHs. Among those TMDCs, MoS₂ is one of the very good semiconductor materials. MoS₂ has a direct band gap of 1.8 eV, a high on/off current ratio of 10⁸, a high carrier mobility of 200 cm² V⁻¹ s⁻¹ at room temperature and outstanding mechanical properties.¹⁸ These properties indicate that 2D monolayer MoS₂ has potential advantages in the fields of nanoelectronics and optoelectronics. On the other hand, the group IIIA monochalcogenides MX (M = Ga or In, X = S, Se, or Te) are emerging 2D materials, which have been aroused great research attention in future optoelectronic devices.^{19–21} Among these diverse MX materials, GaTe monolayer has attracted increasing interest owing to its excellent structural, electronic, optical and photocatalytic properties.^{22,23} The GaTe monolayer has been successfully synthesized by using molecular beam epitaxy or vapor phase deposition methods.^{24,25} Huang *et al.* found that the GaTe

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monolayer with an indirect-band gap is a potential photocatalyst for water splitting, which is a promising solution for generating clean renewable energy.²³ Chitara *et al.* demonstrated that GaTe has good thermoelectric properties and high photo-responsivity (10^4 A W^{-1}), which are better than single-layer graphene.²⁶

vdWHs composed with MoS₂ and GaTe have great potential applications based on their distinct properties. Especially, Wang *et al.* recently experimentally fabricated GaTe/MoS₂ vdWHs by mechanical exfoliation and targeted transfer method, which shows a high photovoltaic and photo-detecting performance.²⁷ Therefore, it is interesting to theoretically understand the physical properties in GaTe/MoS₂ vdWHs. In this work, by using the first-principles calculations, we investigate the physical properties of GaTe/MoS₂ vdWHs, including the geometric structure, electronic and optical properties, *etc.* The structure of the article is as follows: the theoretical calculation methods are shown in Section 2, the results analysis is given in Section 3, and a conclusion is listed in Section 4.

2. Methods

First-principles calculations are performed by using the density functional theory (DFT) approach as implemented in the Vienna *ab initio* Simulation Package (VASP) code with the projector-augmented-wave (PAW) method.^{28–30} For the exchange and correlation energies, the generalized gradient approximation (GGA) in the form of Perdew–Burke–Ernzerhof (PBE) functional is adopted.³¹ The DFT-D2 method advanced by Grimme is used to describe the weak vdW interaction between the two monolayers in all calculations.³² The first Brillouin-zone sampling of $5 \times 5 \times 1$ and $9 \times 9 \times 1$ *k*-points generated by Monkhorst–Pack scheme³³ are used for geometric optimization and optical property calculations, respectively. The vacuum layer is set to 20 Å along the *Z* direction to avoid the interactions induced by the periodic effects. To find a balance between accuracy and efficiency, a 400 eV cut-off energy has been adopted in all calculations. The convergence of the total energy is considered achievable until the energy is less than 10^{-4} eV and the Hellmann–Feynman force on each atom is 0.01 eV Å^{-1} . It should be pointed out that Mo-based and Te-based materials show the strong spin-orbital couplings (SOC),^{34,35} however, we mainly focus on the effects of the strain forces on the band structure and the optical properties in GaTe/MoS₂ heterostructure in this work, so SOC is not included in the following calculations.

In order to study the optical absorption property of GaTe and MoS₂ monolayers as well as the GaTe/MoS₂ vdWH, the dielectric constant formula need to be obtained. The frequency dependent complex dielectric function is formed by adding the real and imaginary parts as follows,

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) \quad (1)$$

The imaginary part $\varepsilon_2(\omega)$ can be obtained by summing up enough empty band states by the following formula,³⁶

$$\varepsilon_2(\omega) = \frac{4\pi^2 e^2}{\Omega} \lim_{q \rightarrow 0} \frac{1}{q^2} \sum_{c,v;k} 2\omega_k \delta(\varepsilon_{ck} - \varepsilon_{vk} - \omega) \times \langle u_{ck+e_\alpha} | u_{vk} \rangle \langle u_{ck+e_\beta} | u_{vk} \rangle^* \quad (2)$$

where Ω stands for the volume, α and β are the Cartesian components, e_α and e_β are the unit vectors, v and c represent matrix elements of the transition from the valence band state (u_{vk}) to the conduction band state (u_{ck}), ε_{ck} and ε_{vk} denote for the energy of the conduction and valence band respectively. The real part $\varepsilon_1(\omega)$ is derived from the Kramers–Kronig equation,³⁷

$$\varepsilon_1(\omega) = 1 + \frac{2}{\pi} P \int_0^\infty \frac{\varepsilon_2^{\alpha\beta}(\omega') \omega'}{\omega'^2 - \omega^2 + i\eta} d\omega' \quad (3)$$

where P represents the principal value of the integral. From these two parts of the dielectric functions, the absorption coefficient $\alpha(\omega)$ can be obtained by using the following formula.³⁸

$$\alpha(\omega) = \frac{\sqrt{2}\omega}{c} \left\{ [\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)]^{\frac{1}{2}} - \varepsilon_1(\omega) \right\}^{\frac{1}{2}} \quad (4)$$

3. Results and discussion

Before studying the properties of the GaTe/MoS₂ vdWH, we first consider the structural and electronic properties of the GaTe and MoS₂ monolayers. The GaTe monolayer has the hexagonal lattice with a repeating unit of Te–Ga–Ga–Te sequence *via* covalent bonds, and the calculated lattice parameters are $a = b = 4.13 \text{ Å}$. The bond lengths of the Te–Ga and Ga–Ga bonds are $d_{\text{Te–Ga}} = 2.71 \text{ Å}$ and $d_{\text{Ga–Ga}} = 2.47 \text{ Å}$, which agree with the previous theoretical and experimental results.²² MoS₂ monolayer also presents hexagonal structures, and the intralayer atoms are sandwiched in the order of S–Mo–S. The lattice constants of MoS₂ are $a = b = 3.17 \text{ Å}$, which agree with the reported literatures.¹⁸ In order to minimize the lattice mismatch value of the vdWH of 1.51%, we utilized a 2×2 GaTe supercell and a $\sqrt{7} \times \sqrt{7}$ MoS₂ supercell to construct the GaTe/MoS₂ vdWH.

To discuss the stacking pattern effects in the GaTe/MoS₂ vdWH, four kinds of possible stacking configurations are considered, which are shown in Fig. 1. Te atom locates at the top site of S atom, Te atom locates at the top site of Mo atom, Ga atom locates at the top site of S atom, Ga atom locates at the top site of Mo atom, respectively. All the configurations are completely relaxed. It is clear that both GaTe and MoS₂ monolayers maintain their original structures without significant deformations. In order to compare the relatively structural stability of GaTe/MoS₂ vdWH, we calculate the binding energies E_b between the GaTe and MoS₂ monolayers from the formula $E_b = E_{\text{total}} - E_{\text{GaTe}} - E_{\text{MoS}_2}$, where E_{total} , E_{GaTe} and E_{MoS_2} are the total energies of GaTe/MoS₂ vdWH, the pristine GaTe and MoS₂ monolayers, respectively. The calculated binding energies for the four different stacking conformations are -13.58 meV , -16.65 meV , -12.32 meV and -16.28 meV , respectively. The calculated results indicate that GaTe and MoS₂ monolayers can form thermodynamically stable GaTe/MoS₂ vdWHs. Among



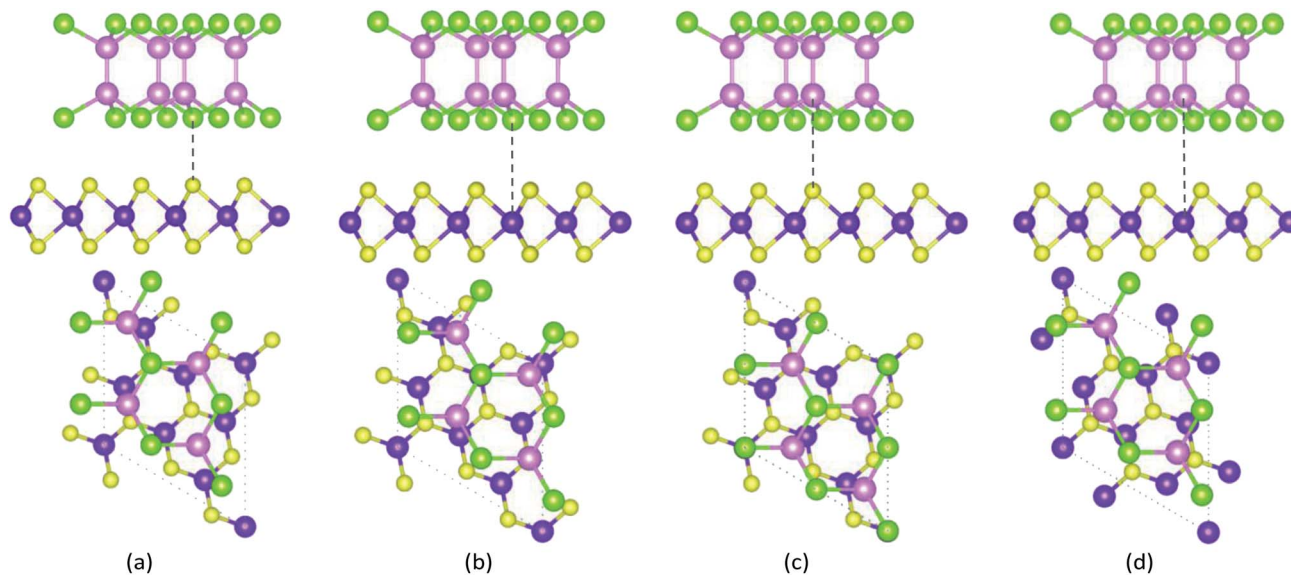


Fig. 1 The top and side views of four stacking patterns of the GaTe/MoS₂ vdWHs. Pink, green, purple and yellow balls represent Ga, Te, Mo and S atoms, respectively.

them, it is found that the binding energy will be the lowest when the Te atom locates at the top site of Mo atom, which means that this stacking pattern is the most stable one. Meanwhile, the optimized equilibrium distance between GaTe and MoS₂

monolayers is 3.56 Å, which is defined as the interlayer height difference between the nearest Te (top-layer) and Mo (lower-layer) atoms. In the following studies, only the most stable configuration is considered.

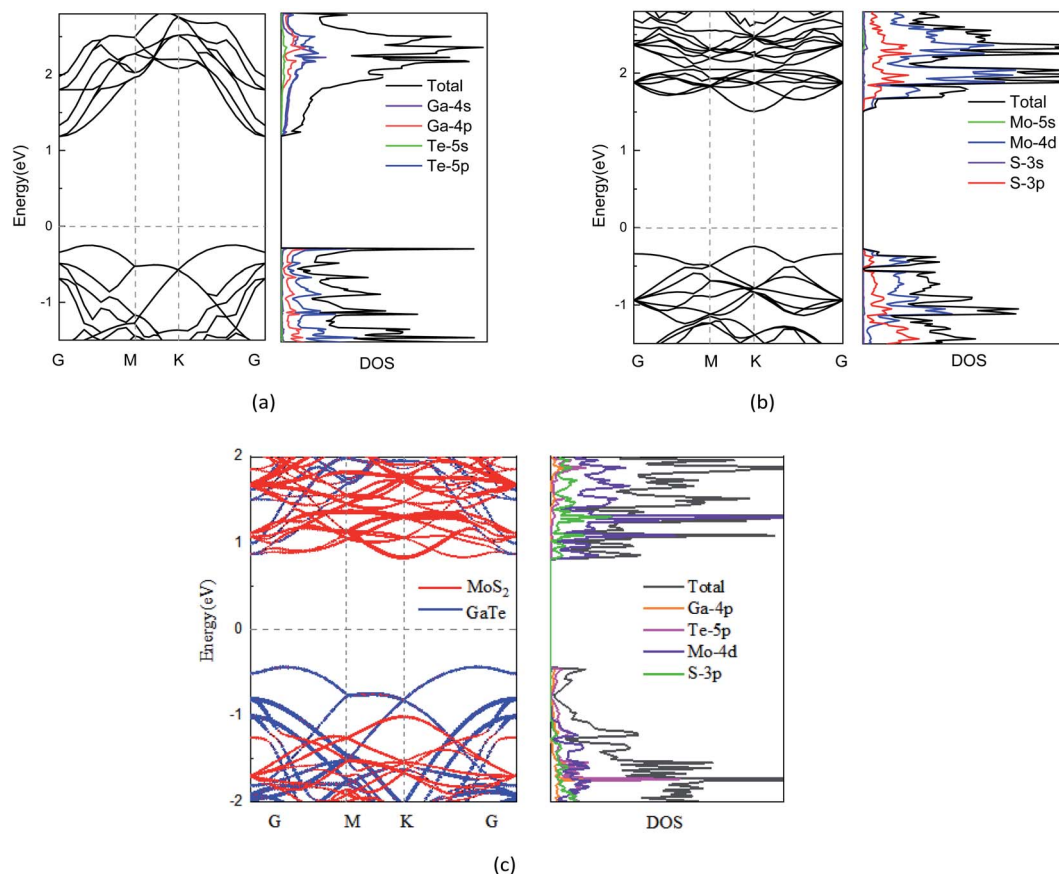


Fig. 2 Band structure and PDOS of (a) 2×2 GaTe monolayer, (b) $\sqrt{7} \times \sqrt{7}$ MoS₂ monolayer, and (c) GaTe/MoS₂ vdWH, respectively.



Fig. 2(a) and (b) present the electronic band structures of the isolated GaTe and MoS₂ monolayers. One can observe that both GaTe and MoS₂ monolayers are semiconductors. The GaTe monolayer, as shown Fig. 2(a), has an indirect band gap of 1.42 eV. Its conduction band minimum (CBM) locates at the *T* point, while its valence band maximum (VBM) locates at the Γ -M path in the two-dimensional hexagonal Brillouin zone. In contrast, the MoS₂ monolayer has a direct band gap of 1.75 eV, which is coming from both the CBM and VBM at the *K* point. After combining GaTe and MoS₂ monolayers together, we plot the band structures of GaTe/MoS₂ vdWH, which are shown in Fig. 2(c). Based on the orbital characteristics, the band structures are plotted by using different colors. It can be seen that the CBM and the VBM of the vdWH are contributed from the MoS₂ and GaTe layers, respectively. Thus, this GaTe/MoS₂ vdWH has an indirect band gap of 1.26 eV with an inherent type-II band alignment, namely, in which both the valence and the conduction band edges of the GaTe monolayer are higher in energy than the corresponding bands of the MoS₂ monolayer. The type-II band alignment can effectively promote the spontaneous separation of the electron-hole pairs and make it a good candidate for various optoelectronics and solar energy conversion devices. Additionally, to get deep insight into the electronic properties of these system, we further examine their density of states (DOS). As depicted in Fig. 2(c), the projected DOS (PDOS) of GaTe/MoS₂ shows that the CBM mainly derives from the Mo-4d and S-3d of MoS₂, whereas the VBM mostly contains the 4p orbitals of Ga and 5p orbitals of Te in GaTe, which further reveal the type-II characteristics of the vdWH.

Fig. 3(a) shows the electrostatic potential of the GaTe/MoS₂ vdWH at equilibrium distance along the *z*-direction in its stacking configurations. It can be seen that there is a large electrostatic potential difference between MoS₂ and GaTe, in which the MoS₂ has a deeper potential than that of the GaTe monolayer. Such different electrostatic potential phenomenon indicates the presence of strong electrostatic field, which may

largely influence the carrier dynamics and charge injection. The difference in electrostatic potential indicates the charge transfers from the higher potential layer to the deeper potential layer. Thus, it is apparent that electrons are likely to be driven from the MoS₂ layer to the GaTe layer, resulting in the enrichment of holes in MoS₂ and the accumulation of electrons in GaTe. To deeper understand the innate character of the charge transfer in the GaTe/MoS₂ vdWH, we also investigate the charge density difference, which is calculated as $\Delta\rho = \rho_{\text{Hetero}} - \rho_{\text{GaTe}} - \rho_{\text{MoS}_2}$. The results are shown in Fig. 3(b). Here ρ_{Hetero} is the total charge density of GaTe/MoS₂ vdWH, ρ_{GaTe} and ρ_{MoS_2} are the charge densities of the GaTe monolayer and MoS₂ monolayer, respectively. The difference in charge density clearly indicates that by forming the electron-rich and hole-rich regions, the charge density is redistributed. And from Fig. 3(b), it can be clearly seen that the electrons transfer from MoS₂ to GaTe, where the red

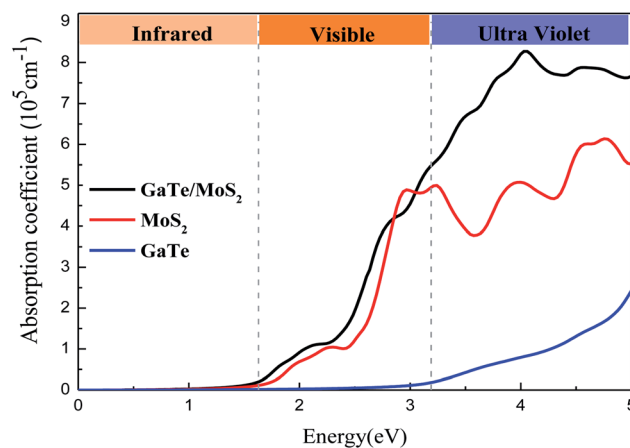


Fig. 4 Adsorption coefficients of the GaTe monolayer, MoS₂ monolayer, and GaTe/MoS₂ vdWH.

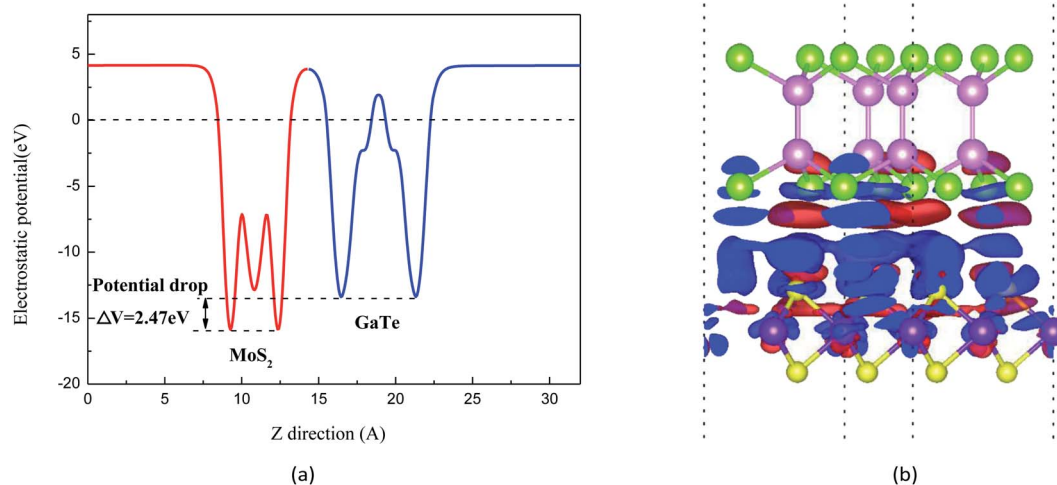


Fig. 3 (a) Electrostatic potential along *z*-direction of the GaTe/MoS₂ vdWH. The red and blue lines represent the MoS₂ and GaTe. (b) Charge density distributions of the GaTe/MoS₂ vdWH at the equilibrium state. The red and blue area represent the electron accumulation and depletion, respectively. The isosurface value is set to be 0.0003 V Å⁻¹.



area represents the accumulation of electrons and the blue area represents the depletion of electrons.

In order to achieve efficient applications in optoelectronic vdWH materials, we calculated the optical properties for the isolated GaTe and MoS₂ monolayers as well as GaTe/MoS₂ vdWH. In Fig. S1 in the ESI,[†] we calculate the dielectric constants $\epsilon(\omega)$ of the system and we found that the GaTe/MoS₂ vdWH has the largest dielectric constant compared with that of the monolayers in the visible region. As shown in Fig. 4, it is noted that both GaTe and MoS₂ monolayers can absorb visible light ($1.6 \text{ eV} < E < 3.1 \text{ eV}$). Meanwhile, it is clear that the optical absorption of the GaTe monolayer is poor and its response to visible light needs to be improved, while the absorption coefficient of MoS₂ monolayer is higher than that of GaTe. Through the combination of GaTe and MoS₂ monolayer, the GaTe/MoS₂ vdWH shows significant enhancement in both visible light and ultraviolet region. And the enhancement of optical absorption can be realized by the fact that the interlayer coupling interaction between the monolayers causes the band gap reducing and the charge transfer. Thus, constructing a vdWH is one of the effective methods to enhance light absorption efficiency. Generally, the GaTe/MoS₂ vdWH has great development potential in optoelectronic devices, especially in the ultraviolet region.

Recently, many studies have shown that applying external strain is an effective approach to tune electronic properties of 2D materials.¹⁷ In order to further understand the influence of the mechanical strain (compressive and tensile strain) of the vdWH on the electronic properties, we apply biaxial strain ($-9\% \leq \epsilon \leq 9\%$) to the system. By changing the lattice parameters, the in-plane biaxial strain on the GaTe/MoS₂ vdWH is simulated and calculated with the formula $\epsilon = [(a - a_0)/a_0] \times 100\%$, where a_0 and a represent the lattice parameters of unstrained and strained of the vdWH, respectively. The variation of the band gaps of the monolayers and GaTe/MoS₂ vdWH with different strains is shown in Fig. 5 and in Fig. S2 in the ESI.[†] It is clearly that for the GaTe/MoS₂ vdWH, when the compressive strain

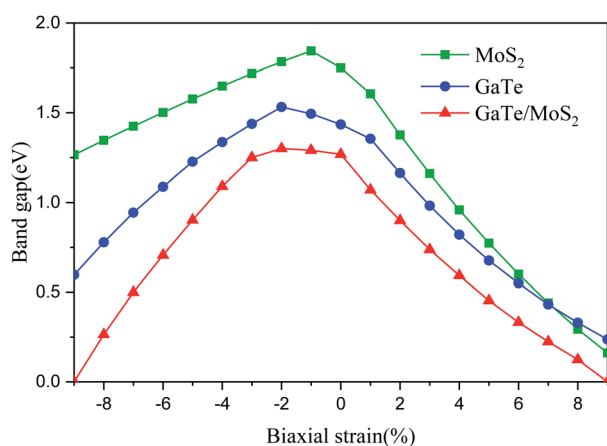


Fig. 5 Effects of strain on the band gaps of GaTe and MoS₂ monolayers and the GaTe/MoS₂ vdWH. Here, positive strain and negative strain represent tensile and compression, respectively.

changes from -9% to -2% , the band gap monotonically increases; when the strain changes from -2% to 9% , the band gap monotonically decreases. And for tensile and compressive strains in excess of 9% , the GaTe/MoS₂ vdWH has undergone a semiconductor-to-metal transition, which means that the vdWH has adjustable conductive and transmission properties. Additionally, the band gap variation of GaTe monolayer and MoS₂ monolayer as a function of strain exhibits a similar trend compared with those in the GaTe/MoS₂ vdWH.

For optical properties, a relative change occurs after applying the biaxial strain. From Fig. 6(a), we can see that the optical absorption coefficients in visible region increase as the tensile strain increases. That is to say, the optical absorption in low energy region widens, especially in the visible regions. In the visible spectrum, the results show that the tensile strain causes a red shift of the spectrum. It is well known that when the optical band gap is small, less energy can be used to excite photoelectrons. In Fig. 6(a), the optical band gap of the GaTe/MoS₂ vdWH is significantly reduced after the strain is applied, and the light absorption is enhanced in the visible light region. This shows that we can use lower energy to excite the photoelectrons in the

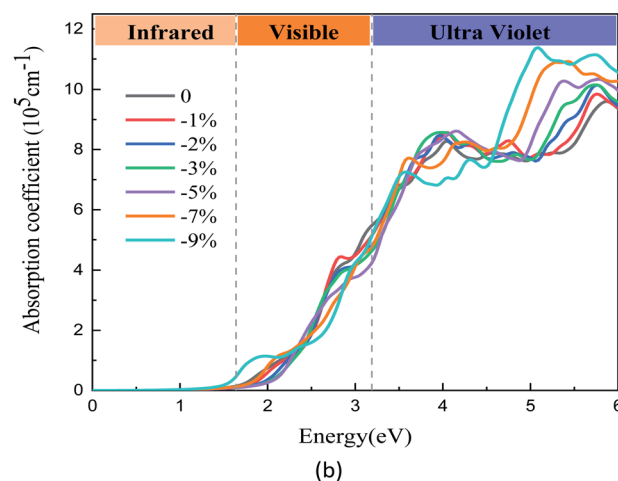
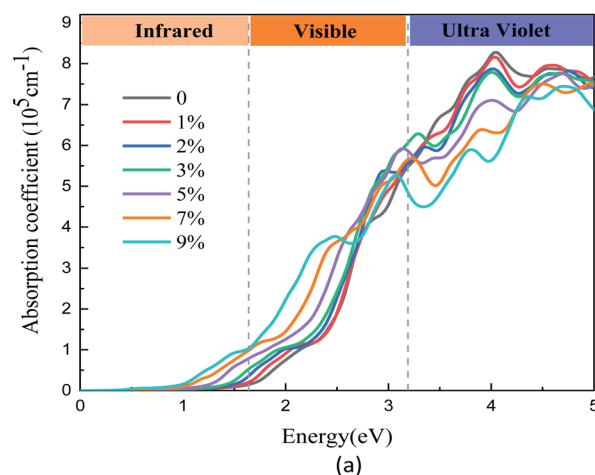


Fig. 6 The optical adsorption spectrum of GaTe/MoS₂ vdWH under (a) tensile and (b) compressive strains.



heterostructure compared to the non-stressed. And similarly, in Fig. 6(b), it can be seen that the compressive strains significantly enhance light absorption in the ultraviolet region. Thus, the tensile and compressive strain have opposite effects on the optical absorption coefficient in the ultraviolet region. The tensile strains increase the optical absorption coefficients in infrared and visible regions, however, the compressive strain increase the optical absorption coefficients in ultraviolet region. Overall, these results indicate that the optical properties of the vdWH can be effectively tuned by biaxial strain.

4. Conclusions

In summary, we have investigated the structural, electronic, and optical properties of GaTe/MoS₂ vdWH through first principles calculations. Our results show that such GaTe/MoS₂ vdWH is characterized by the weak vdW interactions between the GaTe and MoS₂ monolayers. GaTe/MoS₂ vdWH exhibits an indirect band gap semiconductor with type-II band alignment, in which the electrostatic potential and charge difference also indicate the charge transfer from MoS₂ layer to GaTe layer. Simultaneously, the GaTe/MoS₂ vdWH shows enhanced optical absorption properties in the visible light region and the near infrared and UV regions compared with those in the monolayers. In addition, the band gaps and optical properties of the vdWH can be monotonically adjusted through reasonable strains, which indicates that its application prospects in electronic devices exceed other current 2D materials. In particular, large normal strains can make the system change from semiconductor to metal. Also, for optical properties, tensile strain increases the light absorption coefficient in the infrared and visible light regions, while compressive strain increases the light absorption coefficient in the ultraviolet region. These characteristics, tunable electronic structure and optical properties in GaTe/MoS₂ vdWH bring new opportunities for basic research and applications of high-performance nanoelectronic and optoelectronic devices.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666–669.
- 2 M. Chhowalla, H. S. Shin, G. Eda, L. J. Li, K. P. Loh and H. Zhang, *Nat. Chem.*, 2013, **5**, 263–275.
- 3 M. Bernardi, M. Palummo and J. C. Grossman, *Nano Lett.*, 2013, **13**, 3664–3670.
- 4 J. Wang, H. B. Shu, P. Liang, N. Wang, D. Cao and X. S. Chen, *J. Phys. Chem. C*, 2019, **123**, 3861–3867.
- 5 F. F. Zhu, W. J. Chen, Y. Xu, C. L. Gao, D. D. Guan, C. H. Liu, D. Qian, S. C. Zhang and J. F. Jia, *Nat. Mater.*, 2015, **14**, 1020–1025.
- 6 S. Balendhran, S. Walia, H. Nili, S. Sriram and M. Bhaskaran, *Small*, 2015, **11**, 640–652.
- 7 B. Qiu, X. W. Zhao, G. C. Hu, W. W. Yue, X. B. Yuan and J. F. Ren, *Phys. E*, 2020, **116**, 113729.
- 8 C. H. Lee, G. H. Lee, A. M. Van Der Zande, W. C. Chen, Y. L. Li, M. Y. Han, X. Cui, G. Arefe, C. Nuckolls, T. F. Heinz, J. Guo, J. Hone and P. Kim, *Nat. Nanotechnol.*, 2014, **9**, 676–681.
- 9 K. S. Novoselov, A. Mishchenko, A. Carvalho and A. H. Castro Neto, *Science*, 2016, **353**, 9439.
- 10 F. Withers, O. Del Pozo-Zamudio, A. Mishchenko, A. P. Rooney, A. Gholinia, K. Watanabe, T. Taniguchi, S. J. Haigh, A. K. Geim, A. I. Tartakovskii and K. S. Novoselov, *Nat. Mater.*, 2015, **14**, 301–306.
- 11 X. W. Zhao, Z. Yang, J. T. Guo, G. C. Hu, W. W. Yue, X. B. Yuan and J. F. Ren, *Sci. Rep.*, 2020, **10**, 4028.
- 12 B. Qiu, X. W. Zhao, Y. Li, R. D. Liang, G. C. Hu, W. W. Yue, X. B. Yuan and J. F. Ren, *Phys. Lett. A*, 2020, **384**, 126663.
- 13 Y. C. Lin, R. K. Ghosh, R. Addou, N. Lu, S. M. Eichfeld, H. Zhu, M. Y. Li, X. Peng, M. J. Kim, L. J. Li, R. M. Wallace, S. Datta and J. A. Robinson, *Nat. Commun.*, 2015, **6**, 7311.
- 14 M. Massicotte, P. Schmidt, F. Violla, K. G. Schädler, A. Reserbat-Plantey, K. Watanabe, T. Taniguchi, K. J. Tielrooij and F. H. L. Koppens, *Nat. Nanotechnol.*, 2016, **11**, 42–46.
- 15 X. Li, M. W. Lin, J. Lin, B. Huang, A. A. Puzos, C. Ma, K. Wang, W. Zhou, S. T. Pantelides, M. Chi, I. Kravchenko, J. Fowlkes, C. M. Rouleau, D. B. Geohegan and K. Xiao, *Sci. Adv.*, 2016, **2**, 1501882.
- 16 J. Zhang, X. Y. Lang, Y. F. Zhu and Q. Jiang, *Phys. Chem. Chem. Phys.*, 2018, **20**, 17574–17582.
- 17 C. Lei, Y. Ma, X. Xu, T. Zhang, B. Huang and Y. Dai, *J. Phys. Chem. C*, 2019, **123**, 23089–23095.
- 18 M. W. Lin, L. Liu, Q. Lan, X. Tan, K. S. Dhindsa, P. Zeng, V. M. Naik, M. M. C. Cheng and Z. Zhou, *J. Phys. D: Appl. Phys.*, 2012, **45**, 345102.
- 19 Q. Li, Q. Zhou, L. Shi, Q. Chen, J. Wang and A. Huang, *J. Mater. Chem. A*, 2019, **7**, 4291–4312.
- 20 S. Demirci, N. Avazlı, E. Durgun and S. Cahangirov, *Phys. Rev. B*, 2017, **95**, 115409.
- 21 W. Huang, L. Gan, H. Li, Y. Ma and T. Zhai, *CrystEngComm*, 2016, **18**, 3968–3984.
- 22 Y. Guo, S. Zhou, Y. Bai and J. Zhao, *ACS Appl. Mater. Interfaces*, 2017, **9**, 12013–12020.
- 23 A. Huang, W. Shi and Z. Wang, *J. Phys. Chem. C*, 2019, **123**, 11388–11396.



- 24 X. Yuan, L. Tang, S. Liu, P. Wang, Z. Chen, C. Zhang, Y. Liu, W. Wang, Y. Zou, C. Liu, N. Guo, J. Zou, P. Zhou, W. Hu and F. Xiu, *Nano Lett.*, 2015, **15**, 3571–3577.
- 25 F. Liu, H. Shimotani, H. Shang, T. Kanagasekaran, V. Zólyomi, N. Drummond, V. I. Fal'ko and K. Tanigaki, *ACS Nano*, 2014, **8**, 752–760.
- 26 B. Chitara and A. Ya'akovitz, *Nanoscale*, 2018, **10**, 13022–13027.
- 27 F. Wang, Z. Wang, K. Xu, F. Wang, Q. Wang, Y. Huang, L. Yin and J. He, *Nano Lett.*, 2015, **15**, 7558–7566.
- 28 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169.
- 29 G. Kresse and D. Joubert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758.
- 30 P. E. Blöchl, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**, 17953.
- 31 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- 32 S. Grimme, *J. Comput. Chem.*, 2006, **27**, 1787–1799.
- 33 H. J. Monkhorst and J. D. Pack, *Phys. Rev. B: Solid State*, 1976, **13**, 5188.
- 34 J. Wang, H. B. Shu, T. F. Zhao, P. Liang, N. Wang, D. Cao and X. S. Chen, *Phys. Chem. Chem. Phys.*, 2018, **20**, 18571–18578.
- 35 F. A. Rasmussen and K. S. Thygesen, *J. Phys. Chem. C*, 2015, **119**, 13169–13183.
- 36 M. Gajdoš, K. Hummer, G. Kresse, J. Furthmüller and F. Bechstedt, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2006, **73**, 045112.
- 37 T. Eberlein, U. Bangert, R. R. Nair, R. Jones, M. Gass, A. L. Bleloch, K. S. Novoselov, A. Geim and P. R. Briddon, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **77**, 233406.
- 38 B. Luo, X. Wang, E. Tian, G. Li and L. Li, *J. Mater. Chem. C*, 2015, **3**, 8625–8633.

