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Synthesis of indium oxide microparticles using aerosol assisted chemical vapour deposition†

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Microparticles of indium oxide (In_2O_3) are deposited on glass substrates at 500 °C using aerosol assisted chemical vapour deposition (AACVD). The structural, morphological and optical properties of the as-deposited particles are reported.

Indium oxide (In_2O_3) is a promising wide bandgap n-type semiconductor with good optical transparency in the visible region and has drawn significant interest in the field of photovoltaics,^{1,2} thin film transistors,³ photodetectors⁴ and gas sensors.⁵ Doping with other metals (such as Sn, Mo, Zr, Ga, Ti and Ta) is possible in order to modify the properties of In_2O_3 , whilst retaining optical transparency in the visible region of the electromagnetic spectrum. Examples of these widely used doped materials include (i) tin-doped indium oxide (ITO), which is a well known transparent conducting oxide (TCO) has an optical bandgap of 3.4 eV, good chemical stability and excellent adhesion to the substrate, widely used in optoelectronic devices such as light emitting diodes⁶ and solar cells;⁷ (ii) recently, molybdenum-doped In_2O_3 (IMO) has been reported as an durable alternative to the commercially dominant ITO.⁸ IMO has shown higher conductivity and increased infrared transparency than ITO with the same carrier density. This makes IMO a more suitable and low cost alternative material for device applications as much thinner films of IMO can produced with properties equal to or better than ITO of the same carrier concentration; (iii) Zr-doped indium oxide, which has been used as a transparent electrode in perovskite/silicon tandem devices, which results in the improvement of the power conversion efficiency from 23.3% to 26.2%;¹ and (iv) Ga-doped indium oxide which has been used in phase change memory devices.⁹

Several deposition techniques have been used to produce In_2O_3 thin films such as atomic layer deposition,¹⁰ molecular beam epitaxy,¹¹ pulsed laser deposition,¹² spin coating,² metal organic chemical vapour deposition¹³ and aerosol assisted chemical vapour deposition.^{14,15} The latter technique is useful

as it is simple, low cost, single step and used in the industry for assembly line glass coating. Maeng *et al.* reported In_2O_3 films using $\text{Et}_2\text{InN}(\text{TMS})_2$ as a liquid precursor and H_2O in the temperature range of 225–250 °C.¹⁰ Kim *et al.* has used spin coating to deposit In_2O_3 using indium chloride as precursor and annealed at 400 °C.³ In both the cases a poorly crystalline XRD pattern is reported. Basharat *et al.* has reported the deposition of In_2O_3 film at 550 °C on glass substrate from the dual-source AACVD reaction of Me_3In and ROH ($\text{R} = \text{CH}_2\text{CH}_2\text{-NMe}_2$, $\text{CH}(\text{CH}_3)\text{CH}_2\text{NMe}_2$, $\text{C}(\text{CH}_3)_2\text{CH}_2\text{OMe}$, $\text{CH}_2\text{CH}_2\text{OMe}$) in toluene.¹⁴ The chosen ligands are less air/moisture sensitive and have increased solubility, but the preparation of precursor ligands are time consuming (24 h reaction), multi-stepped and require low temperatures (−78 °C). Yang *et al.* reported a one-step aqueous solvothermal method for the synthesis of highly crystalline and nearly monodisperse In_2O_3 nanocrystals.¹⁶ Seo *et al.* reported the preparation of colloidal, highly crystalline and size controlled In_2O_3 nanoparticles from thermal decomposition of the $\text{In}(\text{acac})_3$ precursor in oleylamine.¹⁷ These latter methods yield highly crystalline and monodisperse nanoparticles, but are quite time consuming and not suitable for mass production of In_2O_3 . A comparison table of AACVD and other reported methods for producing indium oxide has been incorporated into the ESI.† The structural phase stability, optical properties, electronic structure and high pressure behaviour of In_2O_3 has been studied by Karazhanov *et al.* using first-principles density functional theory (DFT) calculations in three different space group symmetries $I2_13$, $Ia\bar{3}$ and $R\bar{3}$. It is found that In_2O_3 with space group $Ia\bar{3}$ undergoes a pressure-induced phase transition to the $R\bar{3}$ phase at *ca.* 3.8 GPa. For In_2O_3 the magnitudes of the absorption and reflection coefficients with these space group symmetries are small and in the energy range of 0–5 eV, indicating that these phases are transparent.¹⁸

In this paper we report the deposition of transparent indium oxide microparticles on glass substrates using aerosol assisted chemical vapour deposition (AACVD) from a single precursor solution. AACVD is an ambient pressure CVD technique which is simple, cost-effective, proceeds in a single step and is suitable

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for the production of large area thin films on a range of substrates, and has been used for the deposition of wide range of semiconducting materials such as MAPbBr₃,¹⁹ Cs₂SnI₆,²⁰ MoS₂,²¹ Cr-doped MoS₂,²² SnS²³ and copper zinc tin sulfide (CZTS).²⁴ Solubility of the precursor molecules in a solvent is required in order to obtain high quality thin films. A number of indium precursors have been used to deposit high quality In₂O₃ thin films and whilst current precursor design is functional, none of them are entirely satisfactory and have certain disadvantages,¹⁴ for example, [In(OCMe(CF₃)₂)₃(H₂N^tBu)] and [Me₂-In(OC(CF₃)₂CH₂-NHMe)] contain fluorine, which results in fluorine contamination in In₂O₃ films.^{25,26} [In(thd)₃] (thd = 2,2,6,6-tetramethylheptane-3,5-dionate) was synthesised and added to dichloromethane (CH₂Cl₂), toluene and tetrahydrofuran (THF) separately, but in all cases a fine suspension is formed which, upon standing for 1 h, is sedimented.¹⁵ Indium halides are known for their poor solubility in non-coordinating organic solvents and often rapidly disproportionate to In^{II} or In^{III} halide complexes with treatment of coordinating solvents.²⁷ Keeping the solubility of precursor in mind, we have used for the first time a mixture of polar aprotic solvents to completely dissolve InI to obtain a clear transparent precursor solution for the deposition of In₂O₃ on glass substrates using AACVD.

The precursor solution was obtained by dissolving indium iodide (InI) powder in a mixture of anhydrous *N,N*-dimethylformamide (DMF) and acetonitrile (1 : 1, v/v) with stirring for 1 h at 70 °C. Aerosols were generated from this solution using an ultrasonic humidifier, which were then transported using a stream of argon gas (250 sccm) into a hot wall reactor containing cleaned glass substrates at a temperature of 500 °C. After decomposition of the precursor a transparent film comprised of In₂O₃ microparticles is obtained. The as-deposited materials were characterised using powder X-ray diffraction (p-XRD) in the range of 10° < 2θ < 70° (Fig. 1). Reflections from the (211), (222), (400), (411), (431), (440), (611), (622)

and (622) Bragg planes of cubic In₂O₃ were observed at 2θ = 21.49°, 30.58°, 35.46°, 37.68°, 45.69°, 51.03°, 55.99° and 60.67° respectively (ICCD no. 00-006-0416, space group *Ia*3 with *a* = 10.11 Å). The reflection indexed to the (400) plane is dominant, which indicates a preferred orientation of growth along this direction under these conditions.⁸

The Raman spectrum of the In₂O₃ microparticle film was collected using 514 nm laser excitation. Scattering peaks at 130, 551 and 1371 cm⁻¹ were observed (Fig. 2). The peak around 130 cm⁻¹ is assigned to the A_g⁽¹⁾ vibrational mode and peak at 551 and 1371 cm⁻¹ are ascribed to the phonon vibrational modes of cubic In₂O₃. These values are consistent with those reported in literature.^{28–30} The remaining peaks around 1100 and 1600 cm⁻¹ are assigned to carbon which may arise from decomposition of solvent.³¹

The surface morphology of the as-deposited material was investigated using scanning electron microscopy (SEM) in secondary electron mode. Spherical microparticles are uniformly and randomly distributed over the substrate (Fig. 3(a)). Fig. 3(b) shows the distribution of indium (In) over an interrogated area of ca. 15 × 11 μm confirmed by energy dispersive X-ray (EDX) mapping at an acceleration voltage of 10 kV. Fig. 3(c) shows a histogram which summarises the size distribution of In₂O₃ (*N* = 65). The average diameter of an In₂O₃ particle is 368 ± 120 nm. The mechanisms of indium oxide particle growth and resulting particle sizes have been discussed in literature. Maensiri *et al.* reported that the average particle size of In₂O₃ increases with increasing the calcination temperature.³² Ayeshamariam *et al.* reported that crystallite sizes derived from XRD increases with increasing annealing temperature and that the band gap energy of these particles also scales linearly as a function of annealing temperature. It was not expected to be a quantum confinement effect as the particle sizes studied (ca. 30 nm) were all an order of magnitude greater than the exciton Bohr radius of In₂O₃ (ca. 3 nm).³³ Elemental

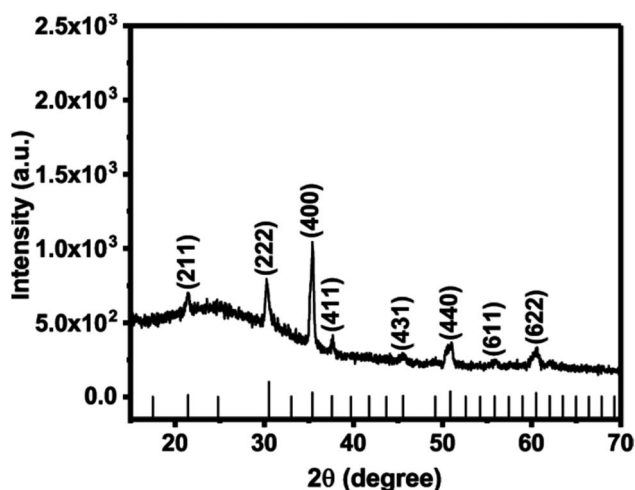


Fig. 1 PXRD pattern of In₂O₃ microparticles deposited by AACVD on glass substrates at 500 °C. The black sticks represent the theoretical powder diffraction pattern of cubic In₂O₃ (ICCD file no. 00-006-0416).

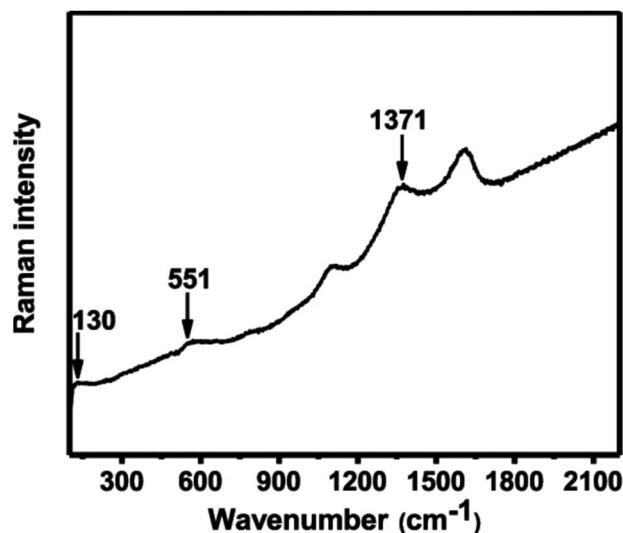


Fig. 2 Raman spectrum of In₂O₃ microparticles deposited by AACVD on a glass substrate at 500 °C.



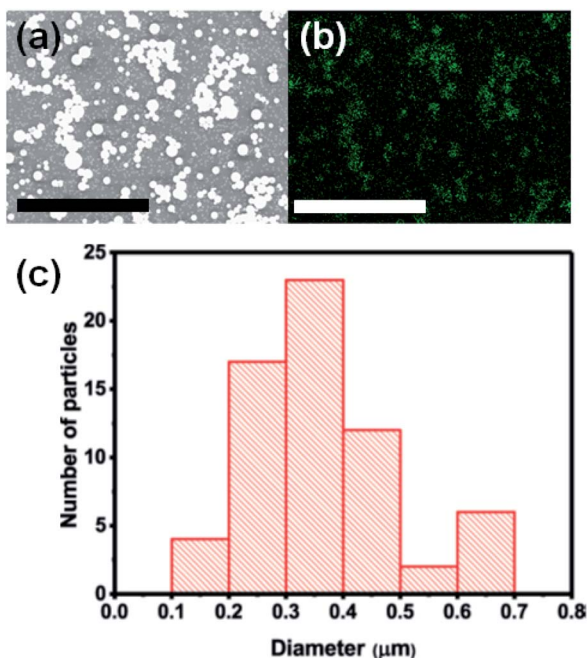


Fig. 3 Characterisation at the microscale using electron microscopy. (a) Secondary electron (SE) SEM image, (b) spatially resolved EDX spectrum mapping of In emission over the same area and (c) histogram of In_2O_3 microparticles deposited by AACVD on a glass substrate at 500°C ($N = 65$). The scale bar in (a) and (b) corresponds to $7\ \mu\text{m}$ in each case.

composition of our In_2O_3 microparticle film was performed using EDX. The EDX spectrum and atomic percent of the elements present in In_2O_3 film are shown in Fig. S1 and S2,[†] giving a clear indication of the presence of In in the film.

The optical properties of as-deposited thin film of In_2O_3 microparticle was investigated by UV-Vis-NIR transmittance in the wavelength range 300–900 nm. The as-deposited In_2O_3 shows over 80% transmittance in the wavelength range of 400–900 nm (Fig. 4), which is comparable to previous reports.¹⁰ The inset of Fig. 4 shows an absorbance spectrum of In_2O_3 microparticles revealing strong optical absorbance in the ultraviolet region of the spectrum. Estimation of optical bandgap of In_2O_3 microparticle thin film from Tauc plot (described in full in the ESI Section S5[†]) gives a direct band gap value of 3.53 eV (Fig. S3[†]). Given the band gap value of 3.53 eV, the In_2O_3 produced by this method and under these conditions is thus a wide band gap semiconductor. By visual inspection it is observed that the In_2O_3 film deposited on glass substrate is colourless and transparent to visible light.^{3,34}

In summary, a simple, cost-effective, and single step aerosol assisted chemical vapour deposition (AACVD) technique has been used for the synthesis of optically transparent spherical In_2O_3 microparticles on glass substrates. Powder XRD patterns and Raman scattering confirm the crystallinity of as-deposited In_2O_3 microparticles with a preferred orientation along the (400) plane. Inspection of surface morphology by secondary electron SEM shows that the as-deposited In_2O_3 microparticles are spherical with an average diameter of $368 \pm 120\ \text{nm}$ ($N = 65$)

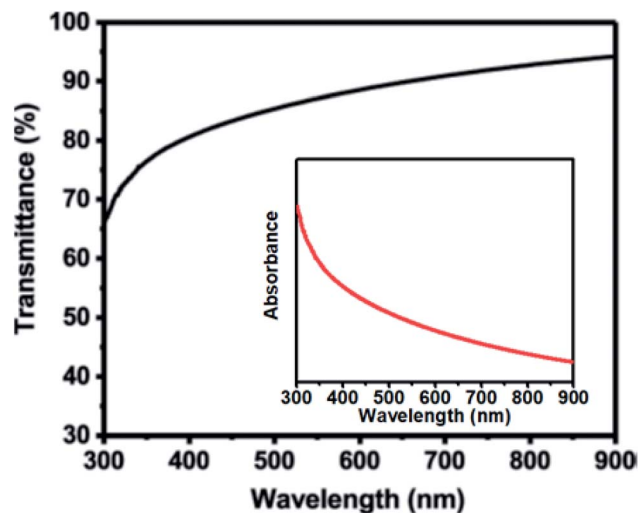


Fig. 4 Optical transmittance spectrum of In_2O_3 microparticles deposited by AACVD technique on a glass substrate at 500°C . Inset: absorbance spectrum of as-deposited In_2O_3 microparticles.

and are uniformly and randomly distributed over the substrate. EDX spectroscopy gives a clear indication of the presence of In on the substrate. The UV-Vis-NIR spectroscopy of as-deposited In_2O_3 microparticle film show over 80% transmittance in the wavelength range of 400–900 nm and strong optical absorbance in the ultraviolet region of the spectrum with an estimated direct bandgap value of 3.53 eV from a Tauc plot. Optical data thus suggest that bandgap value of 3.53 eV does not fall in the band gap range for semiconductors, and hence materials produced by this method are wide band gap semiconductors. Thus the AACVD method we report produces indium oxide from a soluble carbon-free precursor with potential for scale-up.

Conflicts of interest

The authors declare no conflicts of interest.

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Notes and references

- 1 E. Aydin, M. De Bastiani, X. Yang, M. Sajjad, F. Aljamaan, Y. Smirnov, M. N. Hedhili, W. Liu, T. G. Allen, L. Xu, E. Van Kerschaver, M. Morales-Masis, U. Schwingenschlöggl and S. De Wolf, *Adv. Funct. Mater.*, 2019, **29**, 1901741–1901751.
- 2 S. Yoon, S. J. Kim, H. S. Kim, J. S. Park, I. K. Han, J. W. Jung and M. Park, *Nanoscale*, 2017, **9**, 16305–16312.
- 3 H. S. Kim, P. D. Byrne, A. Facchetti and T. J. Marks, *J. Am. Chem. Soc.*, 2008, **130**, 12580–12581.
- 4 S.-Y. Han, G. S. Herman and C.-h. Chang, *J. Am. Chem. Soc.*, 2011, **133**, 5166–5169.



- 5 L. G. Bloor, J. Manzi, R. Binions, I. P. Parkin, D. Pugh, A. Afonja, C. S. Blackman, S. Sathasivam and C. J. Carmalt, *Chem. Mater.*, 2012, **24**, 2864–2871.
- 6 R. Begum, X. Y. Chin, M. Li, B. Damodaran, T. C. Sum, S. Mhaisalkar and N. Mathews, *Chem. Commun.*, 2019, **55**, 5451–5454.
- 7 J.-H. Kim, H.-J. Seok, H.-J. Seo, T.-Y. Seong, J. H. Heo, S.-H. Lim, K.-J. Ahn and H.-K. Kim, *Nanoscale*, 2018, **10**, 20587–20598.
- 8 J. E. Swallow, B. A. Williamson, S. Sathasivam, M. Birkett, T. J. Featherstone, P. A. Murgatroyd, H. J. Edwards, Z. W. Lebens-Higgins, D. A. Duncan and M. Farnworth, *Mater. Horiz.*, 2020, **7**, 236–243.
- 9 S. L. Wang, C. Y. Chen, M. K. Hsieh, W. C. Lee, A. H. Kung and L. H. Peng, *Appl. Phys. Lett.*, 2009, **94**, 113503–113505.
- 10 W. J. Maeng, D.-w. Choi, K.-B. Chung, W. Koh, G.-Y. Kim, S.-Y. Choi and J.-S. Park, *ACS Appl. Mater. Interfaces*, 2014, **6**, 17481–17488.
- 11 O. Bierwagen, M. E. White, M.-Y. Tsai and J. S. Speck, *Appl. Phys. Lett.*, 2009, **95**, 262105–262107.
- 12 E. Tarsa, J. English and J. Speck, *Appl. Phys. Lett.*, 1993, **62**, 2332–2334.
- 13 C. Y. Wang, V. Cimalla, H. Romanus, T. Kups, G. Ecke, T. Stauden, M. Ali, V. Lebedev, J. Pezoldt and O. Ambacher, *Appl. Phys. Lett.*, 2006, **89**, 011904.
- 14 S. Basharat, C. J. Carmalt, S. A. Barnett, D. A. Tocher and H. O. Davies, *Inorg. Chem.*, 2007, **46**, 9473–9480.
- 15 D. Pugh, L. G. Bloor, S. Sathasivam, I. P. Parkin and C. J. Carmalt, *Eur. J. Inorg. Chem.*, 2011, **2011**, 1953–1960.
- 16 J. Yang, C. Li, Z. Quan, D. Kong, X. Zhang, P. Yang and J. Lin, *Cryst. Growth Des.*, 2008, **8**, 695–699.
- 17 W. S. Seo, H. H. Jo, K. Lee and J. T. Park, *Adv. Mater.*, 2003, **15**, 795–797.
- 18 S. Z. Karazhanov, P. Ravindran, P. Vajeeston, A. Ulyashin, T. G. Finstad and H. Fjellvåg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 075129.
- 19 D. J. Lewis and P. O'Brien, *Chem. Commun.*, 2014, **50**, 6319–6321.
- 20 J. C.-R. Ke, D. J. Lewis, A. S. Walton, B. F. Spencer, P. O'Brien, A. G. Thomas and W. R. Flavell, *J. Mater. Chem. A*, 2018, **6**, 11205–11214.
- 21 A. A. Tedstone, D. J. Lewis, R. Hao, S.-M. Mao, P. Bellon, R. S. Averback, C. P. Warrens, K. R. West, P. Howard, S. Gaemers, S. J. Dillon and P. O'Brien, *ACS Appl. Mater. Interfaces*, 2015, **7**, 20929–20934.
- 22 D. J. Lewis, A. A. Tedstone, X. L. Zhong, E. A. Lewis, A. Rooney, N. Savjani, J. R. Brent, S. J. Haigh, M. G. Burke and C. A. Muryn, *Chem. Mater.*, 2015, **27**, 1367–1374.
- 23 P. Kevin, D. J. Lewis, J. Raftery, M. A. Malik and P. O'Brien, *J. Cryst. Growth*, 2015, **415**, 93–99.
- 24 K. Ramasamy, M. A. Malik and P. O'Brien, *Chem. Sci.*, 2011, **2**, 1170–1172.
- 25 L. A. Miinea, S. Suh and D. M. Hoffman, *Inorg. Chem.*, 1999, **38**, 4447–4454.
- 26 T.-Y. Chou, Y. Chi, S.-F. Huang, C.-S. Liu, A. J. Carty, L. Scoles and K. A. Udachin, *Inorg. Chem.*, 2003, **42**, 6041–6049.
- 27 S. P. Green, C. Jones and A. Stasch, *Angew. Chem., Int. Ed.*, 2007, **46**, 8618–8621.
- 28 C. Kranert, R. Schmidt-Grund and M. Grundmann, *Phys. Status Solidi RRL*, 2014, **8**, 554–559.
- 29 K. L. Chitturi, A. Yamma, R. Merugu, R. Dachepalli and J. Kandhadi, *Adv. Nanopart.*, 2016, **5**, 114–122.
- 30 C. K. Latha, M. Raghasudha, Y. Aparna, M. Ramchander, D. Ravinder, K. Jaipal, P. Veerasomaiah and D. Shridhar, *Mater. Res.*, 2017, **20**, 256–263.
- 31 A. Dychalska, P. Popielarski, W. Franków, K. Fabisiak, K. Paprocki and M. Szybowicz, *J. Mater. Sci.*, 2015, **33**, 799–805.
- 32 S. Maensiri, P. Laokul, J. Klinkaewnarong, S. Phokha, V. Promarak and S. Seraphin, *J. Optoelectron. Adv. Mater.*, 2008, **10**, 161–165.
- 33 A. Ayeshamariam, M. Bououdina and C. Sanjeeviraja, *Mater. Sci. Semicond. Process.*, 2013, **16**, 686–695.
- 34 O. Bierwagen, *Semicond. Sci. Technol.*, 2015, **30**, 024001.

