


 Cite this: *RSC Adv.*, 2020, **10**, 18830

Electronic properties and low lattice thermal conductivity (κ_l) of mono-layer (ML) MoS₂: FP-LAPW incorporated with spin-orbit coupling (SOC)

 D. P. Rai,^a Tuan V. Vu,^{b,c} Amel Laref,^d Md. Anwar Hossain,^e Enamul Haque,^e Sohail Ahmad,^f R. Khenata^g and R. K. Thapa^h

This paper focuses on the electronic and thermoelectric properties of monolayer MoS₂. Here, we have examined the structure of MoS₂, in which the hole in the center of the hexagonal cage is considered as a void atom, termed 1H-MoS₂. Density functional theory (DFT) employing the generalized gradient approximation (GGA) and spin-orbit coupling (SOC) has been used for all calculations. Incorporation of SOC resulted in a significant change in the profile of the band energy, specifically the splitting of the valence band maximum (VBM) into two sub-bands. The "split-off" energy is found to be ~20.6 meV. The reduction of the band gap with SOC is a prominent feature at the K-K location in the Brillouin zone. The band gap calculated with the GGA is ~1.75 eV. However, on implementation of SOC, the GGA band gap was reduced to ~1.68 eV. The frequency-dependent phonon dispersion curve was obtained to analyse the thermodynamical stability. 1H-MoS₂ is found to be thermodynamically stable with no imaginary frequency. We report a low value of lattice thermal conductivity (κ_l) and low electron effective masses, which are desirable for potential applications in thermoelectric devices.

Received 20th March 2020

Accepted 20th April 2020

DOI: 10.1039/d0ra02585b

rsc.li/rsc-advances

1 Introduction

The non-existence of an energy band gap in graphene has seriously hindered its technological applications in digital electronics and other low-power devices.¹ The technological limitations of graphene are highly challenging and have forced researchers to look for potential 2D materials with finite band gaps.^{2,3} Ultrathin 2D layered materials like single-layer transition metal dichalcogenides (TMDs) (Mo/WX₂, X = S, Se, Te) with two-fold valley degeneracy are promising due to their appreciable band gap which depends on the thickness and the fact that they exhibit outstanding mechanical properties like those of graphene. TMDs have many industrial applications, such as lubricants,⁴ photo-catalysis,⁵ photo-voltaics⁶ and energy storage.⁷ In particular, MoS₂ is an important TMD due to its availability and

room temperature stability. Furthermore, this material can be synthesized without much effort *via* various experimental techniques, such as chemical vapor deposition,^{8,9} micro-exfoliation¹⁰ or solvent-based technology.^{11,12} Multi-layer MoS₂ has a broad commercial application as a dry lubricant which is due to the weak interlayer van der Waals (vdW) interactions between the adjacent layers.¹³ Bulk MoS₂ is an indirect band gap semiconductor with an energy band gap of ~1.23 eV, while single layer MoS₂ exhibits a direct band gap of ~1.8 eV.^{14,15} The size-dependent tunability of the electronic properties makes MoS₂ a novel material for nanoscale field-effect transistors and optical sensors.^{16–19} Recently, a hetero-junction layer structure of MoS₂-HfO₂ was successfully constructed and implemented in a nanoscale field-effect transistor (NFET).²⁰ A WS₂/MoS₂ heterojunction was mechanically fabricated and the room temperature photoluminescence spectra were studied by theory and experiments.²¹ Several theoretical studies are in progress, focusing on the physical and chemical properties of 2D MoS₂ under different applied fields using an *ab initio* approach.^{22–25} Besides the above mentioned functional properties, TMDs have interesting thermoelectric properties which can be utilized in emerging energy harvest applications.^{26–29} Thermoelectric materials have potential technological importance in converting industrial waste heat into electrical energy and *vice versa*.^{30–33} Several studies have reported that TMDs could be potential thermoelectric materials due to their low lattice thermal conductivities and high charge mobilities due to their small effective masses, but the benchmark performance is still in the nascent stage in terms of

^aPhysical Sciences Research Center (PSRC), Department of Physics, Pachhunga University College, Aizawl-796001, India

^bDivision of Computational Physics, Institute for Computational Science, Ton Duc Thang University, Ho Chi Minh City, Vietnam. E-mail: vuvantuan@tdtu.edu.vn

^cFaculty of Electrical & Electronics Engineering, Ton Duc Thang University, Ho Chi Minh City, Vietnam

^dDepartment of Physics, College of Science, King Saud University, Riyadh, Saudi Arabia

^eDepartment of Physics, Faculty of Science, Mawlana Bhashani Science and Technology University, Santosh, Tangail-1902, Bangladesh

^fDepartment of Physics, Faculty of Science, King Khalid University, Abha, Saudi Arabia

^gLaboratoire de Physique Quantique de la Matière et de la Modélisation Mathématique (LPQ3M), Faculté des Sciences, Université de Mascara, Mascara 29000, Algeria

^hDepartment of Physics, Mizoram University, Aizawl-796004, India


practical applications.^{34–38} The thermoelectric performance of a solid-state material depends on the dimensionless figure of merit called thermoelectric efficiency (ZT), calculated as

$$ZT = \frac{S\sigma^2}{\kappa_1 + \kappa_e} T \quad (1)$$

where S is the Seebeck coefficient, σ is the electrical conductivity, κ_e is the electronic thermal conductivity and κ_1 is the lattice thermal conductivity. A good thermoelectric material possesses high S , high σ and low thermal conductivity, $\kappa = \kappa_e + \kappa_1$. Narrow band gap (0.5–1.8 eV) semiconductors are preferable, and MoS₂ looks promising in this regard and could have a high ZT value. Recently, a high power factor value ($P = S\sigma^2$) of ~ 8.5 mW m⁻¹ K⁻¹ has been experimentally reported in few-layer MoS₂ at room temperature.³⁹ On the other hand, the thermoelectric efficiency (ZT) of single layer MoS₂ is very low (0.11 at 500 K).⁴⁰ Huang *et al.*,⁴¹ using a ballistic model, reported an improved ZT value of up to 0.5 (300 K). Other monolayer TMDs like PdS₂,³⁸ MoSe₂,⁴² WSe₂,⁴² WS₂ (ref. 43) and SnSe₂ (ref. 44) also show significant thermoelectric response with ZT values of 0.1 (1200 K), 0.8 (1200 K), 0.90 (1500 K), 1.1 (300 K) and 2.95 (800 K), respectively. However, on structural modification the ZT value has been enhanced up to 3.5 at 300 K for armchair nanoribbons (ACNRs), and also bilayer MoSe₂ shows a maximum room temperature ZT value of ~ 2.4 .^{34,40} Arab *et al.* have also reported high values of $ZT = 4.0$ in 3-ACNRs for n-type MoS₂ and $ZT = 3$ in 4-ACNRs for p-type MoS₂ at 500 K.⁴⁵ As we know, ZT is inversely related to the thermal conductivity ($\kappa = \kappa_e + \kappa_1$). For instance, monolayer ZrS₂, ZrSe₂ and HfSe₂ exhibit low κ_1 values of 3.29,⁴⁶ 1.2 and 1.8 W m⁻¹ K⁻¹,⁴⁷ respectively at 300 K, and the respective ZT values are 1.65, 0.87 and 0.95. Even smaller values of κ_1 have been reported for disordered 2D-MoS₂ and 2D-WSe₂, *i.e.* 0.05 W m⁻¹ K⁻¹ and 0.1–1 W m⁻¹ K⁻¹,^{48,49} respectively. Moreover, the κ_1 value of MoS₂ nanoribbons was found to be ~ 5 W m⁻¹ K⁻¹ at room temperature using molecular dynamics (MD) theory.⁵⁰ Different values of κ_1 have been reported for 2D-MoS₂ based on different approaches, *viz.* 1.35 W m⁻¹ K⁻¹, 23.2 W m⁻¹ K⁻¹ and 26.2 W m⁻¹ K⁻¹ obtained from molecular dynamics (MD),⁵¹ a non-equilibrium Green's function,⁵² and the Boltzmann semi-classical transport equation (BTE),³⁶ respectively. However, some articles have reported high values of κ_1 ; 83 W m⁻¹ K⁻¹,⁵³ 52 W m⁻¹ K⁻¹ (ref. 35) and 85–110 W m⁻¹ K⁻¹ (ref. 54) were observed for the vapour phase of few-layer MoS₂ and the (001) orientation of a MoS₂ crystal with basal-plane thermal conductivity as a function of laser spot size. The non-uniformity of thermoelectric responses arises due to the variation in the values of κ_1 . Recently, Kaur *et al.*, using the semi-classical Boltzmann transport approach, reported low values of κ_1 of around 8.3 and 5 W m⁻¹ K⁻¹ for 2D ScP and ScAs, respectively.⁵⁵ A large value for the Seebeck coefficient of -4×10^2 to -1×10^5 μ V K⁻¹ has also been reported in monolayer MoS₂.⁵⁶ It seems that the studies of the thermoelectric behaviour of MoS₂ are still very crude. Therefore, we need a more rigorous and accurate study for the concrete determination of the κ_1 value of MoS₂. In this paper, an investigation has been carried out on the thermoelectric response of 2D 1H-MoS₂ using the first principles method incorporating spin-orbit coupling

(SOC) along with the semi-classical Boltzmann Transport Equation (BTE) as implemented in BoltzTraP code.⁵⁷

2 Computational details

The electronic and phonon properties are computed based on Kohn–Sham density functional theory (KS-DFT) using the two computational packages WIEN2K⁵⁸ and QUANTUM ESPRESSO,⁵⁹ respectively. WIEN2K relies on the full-potential linearized augmented plane wave (FP-LAPW) method, whereas QUANTUM ESPRESSO incorporates an ultrasoft pseudopotential. A generalized gradient approximation (GGA) developed by Perdew–Burke–Ernzerhof (PBE)⁶⁰ has been considered for electron exchange–correlation. The valence and semi-core state electrons are treated relativistically considering the spin–orbit coupling (SOC). A dense optimized $10 \times 10 \times 1$ k -mesh is adopted for the first Brillouin zone integration in which 286 irreducible k -points are used for the energy calculations. The convergence criterion for the complete self-consistency calculation is set as 0.0001 Ry. Bulk MoS₂ crystallizes in a hexagonal structure with point group $6/mmm$, space group $P6_3/mmc$ and lattice constants $a = b = 3.19$ Å, $c/a = 3.86$ Å.⁶¹ A 2D slab was constructed by taking a 10 Å vacuum along the z -axis with a void in the center of the hexagonal ring. The 2D slab was optimized by taking the minimum energy until the self-consistent calculation reached below the Hellmann–Feynman force of 0.01 Ry Å⁻¹. The optimized lattice constant was found to be $a = b = 3.185$ Å and agrees well with the previous finding of 3.183 Å.⁶² The hexagonal crystal structure of monolayer MoS₂ (both top and side views) is depicted in Fig. 1. To determine the phonon spectrum of MoS₂, we used a $16 \times 16 \times 1$ k -mesh and a $2 \times 2 \times 1$ q -mesh. The electron and lattice parts of the thermoelectric parameters were calculated with a $16 \times 16 \times 1$ k -mesh by using the first-principles Boltzmann semi-classical transport equation with a single-mode relaxation-time approximation called BoltzTraP⁵⁷ and Phono3py.⁶³ BoltzTraP is used to calculate the electron part of the thermoelectric parameters and Phono3py determines the lattice thermal conductivity (κ_1).

3 Results and discussion

3.1 Electronic properties

Top and side views of the 2D structure of MoS₂ are presented in Fig. 1. The lattice arrangement of MoS₂ shows clear bonds

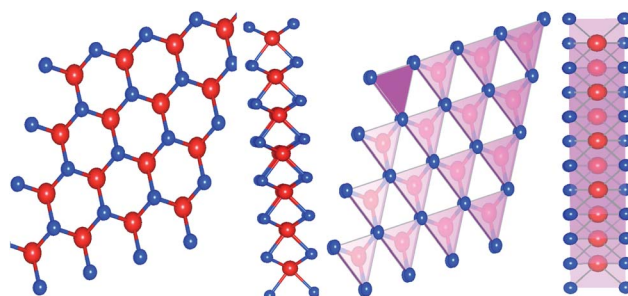


Fig. 1 Top view and side view of the 2D structure of 1H-MoS₂ (Mo – red and S – blue with a polyhedral cage).



between the S and Mo atoms, keeping a void or hollow in the central part. Therefore, the top view exactly resembles a 2D graphene-like structure. Six S atoms and one Mo atom form two symmetrical pyramidal polyhedrons, and the Mo atom is at the junction of the two pyramids. The presence of bonding between the S and Mo atoms may be the reason for p-d hybridization which leads to metallic bonding and the absence of van der Waals interactions. The electronic properties are investigated by calculating the density of states (DOS) and band structure of MoS₂ as shown in Fig. 2 and 3. Based on both (GGA and SOC) calculations, MoS₂ exhibits a clear band gap at the Fermi level (E_F), showing the existence of semiconducting behaviour. The maxima and minima of the dispersed bands are observed at the high symmetry K point. The probability of electron transitions along the K-K symmetry points indicates that MoS₂ is a direct band gap semiconductor. The origin of the energy band gap is due to the Mo-d orbital and S-p orbital hybridization, as discussed elsewhere.^{14,61,64-67} For further elucidation of the band structure we have calculated the partial density of states (PDOS) as well. Fig. 2(a and b) display the partial DOS calculated with GGA and GGA+SOC, respectively. A comparison of the total and partial DOS calculated with GGA and SOC is also presented in Fig. 2(c). Fig. 2(a and b) are divided into three layers: top (total DOS), middle (partial DOS of Mo atoms) and bottom (partial DOS of S atoms). The PDOS of the S atoms represents the 3p states. The first relative magnitude of the 3p states of the S atoms in monolayer MoS₂ (0.31 states per eV) is the same for both GGA and GGA+SOC calculations [Fig. 2(a and b)]. The above results obtained from Fig. 2 and 3 indicate that monolayer MoS₂ is a direct band gap semiconductor with electron transitions along the K-symmetry points. The major contribution is attributed to Mo-d and S-p states. The valence band (VB) in the range from 0 to -6 eV is an admixture of both Mo-d and S-

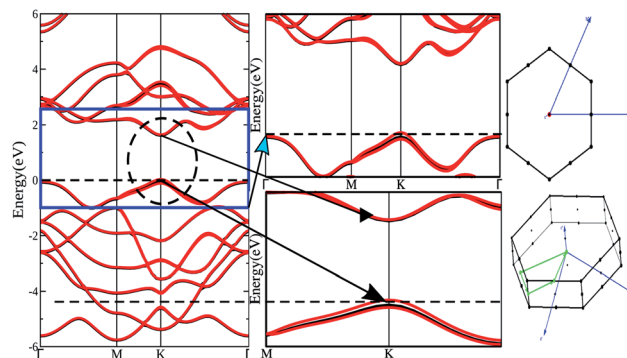


Fig. 3 Band structure of 1H-MoS₂ (GGA: black lines and GGA+SOC: red lines), first Brillouin zone, and primitive cell (blue arrows indicate reciprocal lattice vectors, green lines represent the high symmetry points).

p states due to the p-d orbital hybridization [Fig. 2(a)]. Whereas, in the conduction band (CB), the energy range from 1.75 to 5 eV is mainly composed of occupied Mo-d states with a small contribution from the S-p states. From Fig. 3 (right side), we can see the significant effect of GGA+SOC on the electronic band structure with the splitting of the valence band maximum (VBM). We have observed that the completely filled Mo-d_{z²} state lies at the VBM (Fig. 3). Meanwhile on implementation of GGA+SOC, the occupied Mo-d_{z²} state is pushed towards lower energy while the Mo-d_{x²-y², xy} band is ~0.026 eV higher in energy on approaching the E_F [Fig. 3 (middle bottom line marked by an arrow head) and Fig. 4]. This “split-off” energy is almost half of the experimentally measured value of 0.042 eV.⁶⁸ Also, the unoccupied Mo-d_{z²} state at the CBM drops down by a small amount of energy as compared to the GGA band, as shown in Fig. 2(b). Thus, GGA+SOC reduces the GGA band gap from

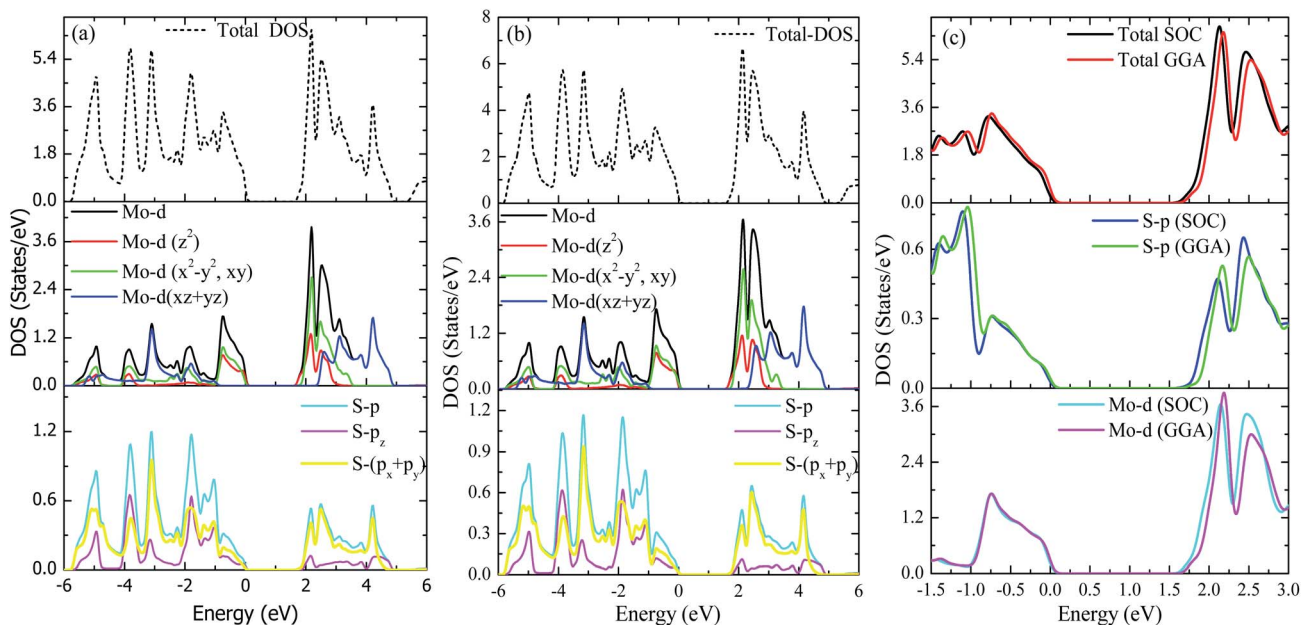


Fig. 2 Total and partial DOS of MoS₂. (a) GGA, (b) GGA+SOC and (c) both GGA and GGA+SOC together.



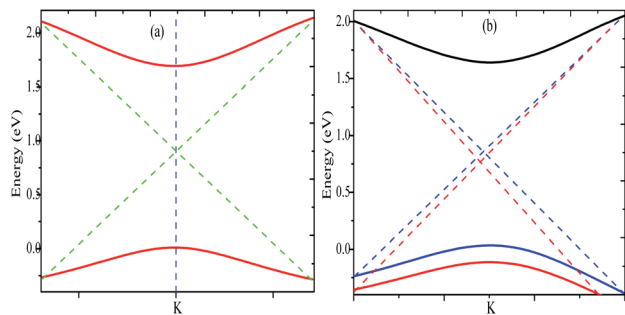


Fig. 4 Valence band maxima and conduction band minima calculated with (a) GGA and (b) GGA+SOC.

~ 1.75 eV to ~ 1.68 eV. The presence of two degenerate bands at the VBM along the K-symmetry point on the application of SOC is probably due to the Mo- $d_{x^2-y^2,xy}$ and Mo- d_{z^2} states.

3.2 Thermoelectric properties

To confirm the thermodynamical stability for practical synthesis, we have calculated the frequency-dependent phonon dispersion using density functional perturbation theory (DFPT)^{63,69,70} as shown in Fig. 5. The presence of phonon modes in the positive frequency range indicates that monolayer MoS₂ is dynamically stable for laboratory synthesis. The three atoms in the primitive cell give nine distinctive vibrational modes along the Γ -point. The nine phonon branches are mixtures of three acoustic (lower frequency) and six optical (higher frequency) branches. As shown in Fig. 5, the acoustic modes are identified as the transverse mode (TA), longitudinal mode (LA), and out-of-plane mode (ZA) whereas the optical branches are composed of two transverse modes (TO) at the bottom, two longitudinal modes (LO) in the middle and two out-of-plane modes (ZO) at the top. The optical mode (TO) and acoustic mode (LA) are well separated by ~ 52 cm⁻¹. The finite frequency band gap as a result of LA-TO splitting may be

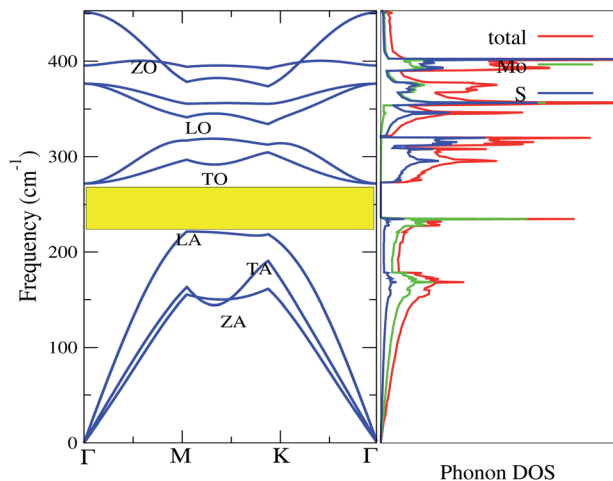


Fig. 5 Phonon bands and phonon DOS of 2D monolayer MoS₂.

attributed to the large difference in the atomic masses (Mo = 95.96 amu and S = 32.06 amu). The optical branches along the Γ symmetry points are at ~ 275 cm⁻¹, ~ 370 cm⁻¹, ~ 395 cm⁻¹ and ~ 460 cm⁻¹, consistent with the previous results.^{37,62} Fig. 6 demonstrates the phonon transport related parameters: (a) group velocity at 300 K, (b) Grüneisen parameter at 300 K and (c) relaxation time (τ). The three acoustic modes have the highest group velocity and Grüneisen parameter at a particular frequency and q -point. This indicates high phonon-phonon scattering rates in which most of the heat is transported by the transverse acoustic mode (TA) and longitudinal acoustic mode (LA).^{71,72} Therefore, the lattice thermal conductivity can be further reduced by doping with the dissipation of heat that may be introduced by the acoustic-optical phonon scattering channels.⁷³ The group velocity of the ZA mode can reach ~ 14 km s⁻¹ at a particular phonon frequency. However, the observation of low group velocity at higher frequencies for the optical modes indicates a small but significant contribution to the thermal conductivity. Anharmonicity results in enhanced phonon-phonon scattering, which reduces κ_l without affecting the electronic properties.⁷⁴ The Grüneisen parameter measures the strength of the anharmonicity. Therefore, the larger the Grüneisen parameter, the stronger the phonon scattering. The high value of the Grüneisen parameter suggests high anharmonicity in 2D-MoS₂ and intense phonon scattering. The short phonon relaxation time of 2D-MoS₂ also indicates intense phonon scattering.

The description of electron transport in a solid-state material is directly related to the electronic energy bands which give rise to a thermoelectric response measured in terms of a dimensionless figure of merit ZT as already presented in eqn (1). As reported earlier, monolayered MoS₂ is a semiconductor with a direct band gap value of ~ 1.8 eV.^{14,64,67} In semiconductors, the effective mass (m^*) plays a vital role in deriving the quantitative transport characteristics. The effective mass of a charge carrier can be obtained from the parabolic band near E_F based on the energy-independent scattering approximation^{79,80} given by eqn (2).

$$m^* = \hbar^2 \left(\frac{\partial^2 E}{\partial k^2} \right)^{-1} \quad (2)$$

The effect of SOC leads to splitting of the band near E_F close to the VBM which results in two effective masses for holes [cf. Fig. 3 and 4]. A lower value of effective mass leads to a higher value for the charge mobility. On the other hand, the electrical conductivity (σ) is directly related to the charge carrier mobility via eqn (3)

$$\sigma = ne\mu \quad (3)$$

where n is the concentration of charge carriers and e is the electron charge. The calculated effective masses for electrons (m_e^*) and holes (m_h^*) are presented in Table 1. Our calculated values of the effective masses are $m_e^* = 0.391$ (GGA), $m_h^* = 0.552$ (GGA), $m_e^* = 0.315$ (SOC) and $m_h^* = 0.467$ (SOC); these values are



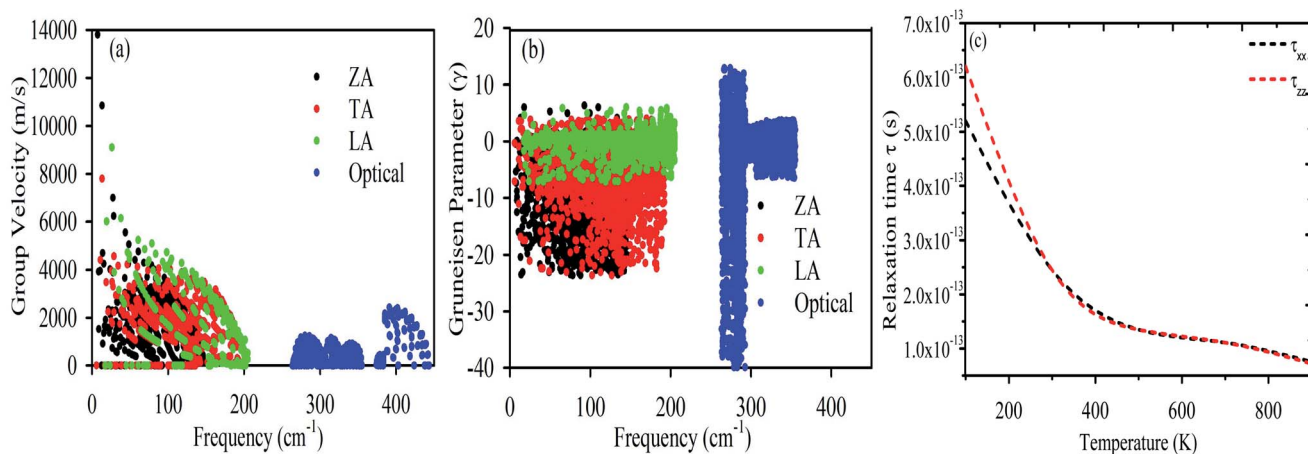


Fig. 6 (a) Group velocity, (b) Grüneisen parameter and (c) phonon relaxation time (τ_p).

in good agreement with the earlier results estimated with the GGA,⁷⁵ DFT-HSE⁷⁸ and tight binding (TB)⁷⁸ approaches. The lattice part of the thermal conductivity κ_l has been determined by using the Phono3py code⁶³ which iteratively solves the BTE from eqn (4):

$$\kappa_l^{\alpha\beta} = \frac{1}{Nk_B T^2 \Omega} \sum_{\lambda} f_0(\omega_{\lambda}) f_0(\omega_{\lambda} + 1) (\hbar\omega)^2 v_{\lambda}^{\alpha} v_{\lambda}^{\beta} \tau_{\lambda} \quad (4)$$

where Ω is the primitive cell volume, α and β are the Cartesian components, k_B is the Boltzmann constant, ω_{λ} and v_{λ} are the angular frequency and group velocity, τ_{λ} denotes the phonon relaxation time and f_0 is the Bose–Einstein distribution function near E_F .

The electron parts of the thermoelectric parameters are given by eqn (5)–(9) as presented in the BoltzTraP user manual.⁵⁷

Here $\sigma_{\alpha,\beta}$ are the electrical conductivity tensors,

$$\sigma_{\alpha,\beta} = e^2 \sum_{i,k} \left[-\frac{\partial f_0(T, \varepsilon, \mu)}{\partial \varepsilon} \right] v_{\alpha} v_{\beta} \tau_k \quad (5)$$

where α and β refer to the tensor indices, v_{α} and v_{β} are the group velocities, e is the charge of the electron and τ_k is the electron relaxation time. The contributions of the electrons are mostly found near the Fermi energy (E_F), termed as the chemical potential (μ) ($\mu - k_B T < \varepsilon < \mu + k_B T$) where k_B is the Boltzmann constant. The kernel of the transport distribution is given by:

$$\Xi_{i,k} = \sum_{i,k} v_{\alpha} v_{\beta} \tau_k \quad (6)$$

The electron relaxation time (τ) dependent electrical conductivity (σ/τ), thermal conductivity (κ/τ) and Seebeck coefficient can be written as

$$\sigma/\tau = e^2 \int \Xi_{i,k} \left[-\frac{\partial f_0(T, \varepsilon, \mu)}{\partial \varepsilon} \right] d\varepsilon \quad (7)$$

$$\kappa_e/\tau = k_B^2 T \int \Xi_{i,k} \left(\frac{\varepsilon - \mu}{k_B T} \right)^2 \left[-\frac{\partial f_0(T, \varepsilon, \mu)}{\partial \varepsilon} \right] d\varepsilon \quad (8)$$

$$S = \frac{ek_B}{\sigma} \int \Xi_{i,k} \left(\frac{\varepsilon - \mu}{k_B T} \right) \left[-\frac{\partial f_0(T, \varepsilon, \mu)}{\partial \varepsilon} \right] d\varepsilon \quad (9)$$

where f_0 is a Fermi–Dirac distribution function.

The relaxation time (τ_e) dependent electrical conductivity (σ), Seebeck coefficient (S), electron thermal conductivity (κ_e) and power factor (PF) as a function of chemical potential (eV) along the x - and z -axes are shown in Fig. 7(a and b), respectively. The electron parts of the thermoelectric parameters are strongly dependent on the electron relaxation time (τ_e). The thermoelectric efficiency (ZT) value cannot be estimated unless and until τ_e is decoupled. Therefore, we have performed the electron transport

Table 1 Calculated energy band gap (E_g) in eV and effective masses (m^*) of electrons (m_e^*) and holes (m_h^*) in terms of electron mass (m_e)

Functional	Energy band gap				m^*		
	CBM	VBM	E_g	Prev. E_g	m_e^*	m_h^*	m_{prev}^*
GGA	−0.02	1.73	1.75	1.77 (ref. 75) 1.77 (ref. 76) 1.78 (ref. 77) 1.786 (ref. 78)	0.391	0.552	$m_h^* = 0.60, m_e^* = 0.46,^{75} m_e^* = 0.54$ (ref. 77)
GGA-SOC	0.00	1.68	1.68		0.315	0.467	$m_h^* = 0.485/0.463, m_e^* = 0.407/0.430$ (ref. 78) $m_h^* = 0.49, m_e^* = 0.44$ (ref. 76)



calculation by using the modified BoltzTraP code based on the electron carrier relaxation time. The expression for electron relaxation time is given by eqn (10)^{82,83}

$$\tau_e^{-1}(\varepsilon, \mu, T) = \frac{2\pi\Omega}{g_s\hbar} \sum_{\nu} g_{\nu}^2(\varepsilon, \varepsilon + \bar{\omega}_{\nu}) [n(\bar{\omega}_{\nu}, T) + f(\varepsilon + \bar{\omega}_{\nu}, \mu, T)] \rho(\varepsilon + \bar{\omega}_{\nu}) + g_{\nu}^2(\varepsilon, \varepsilon - \bar{\omega}_{\nu}) [n(\bar{\omega}_{\nu}, T) + 1 - f(\varepsilon - \bar{\omega}_{\nu}, \mu, T)] \rho(\varepsilon - \bar{\omega}_{\nu}) \quad (10)$$

where Ω refers to the volume of the primitive cell, \hbar is Planck's constant, ν is the phonon mode index, $\bar{\omega}_{\nu}$ is the averaged phonon mode energy, g_{ν}^2 is the averaged electron-phonon matrix, $n(\bar{\omega}_{\nu}, T)$ is the Bose-Einstein distribution function, $f(\varepsilon + \bar{\omega}_{\nu}, \mu, T)$ refers to the Fermi-Dirac distribution function, $g_s = 2$ is the spin degeneracy, ε is the electron energy, and ρ is the density of states per unit energy and unit volume (V). The electron relaxation times along both the x - and z -axes as a function of chemical potential and absolute

temperature are presented in Fig. 8(a-d). The electron relaxation times along the z -axis are longer than those along the x -axis for both n -type and p -type carriers. However, the electron relaxation times of n -type and p -type carriers along the same axis are almost the same. The electrons in the conduction band which lies close to the Fermi energy shows longer relaxation times as compared to the holes in the valence bands along the x -axis. This is in contrast to the carriers along the z -axis. For both carriers, τ_e decreases with an increase in absolute temperature. The sharp decrease in τ_e near the band edges can be visualised as eqn (11)⁸²

$$\tau_e^{-1} = \sim g^2(\varepsilon)\rho(\varepsilon) \quad (11)$$

Eqn (11) established an inverse relation between τ_e and the carrier density of states per unit energy and volume (ρ), while the electron-phonon matrix elements (g) depend weakly on the

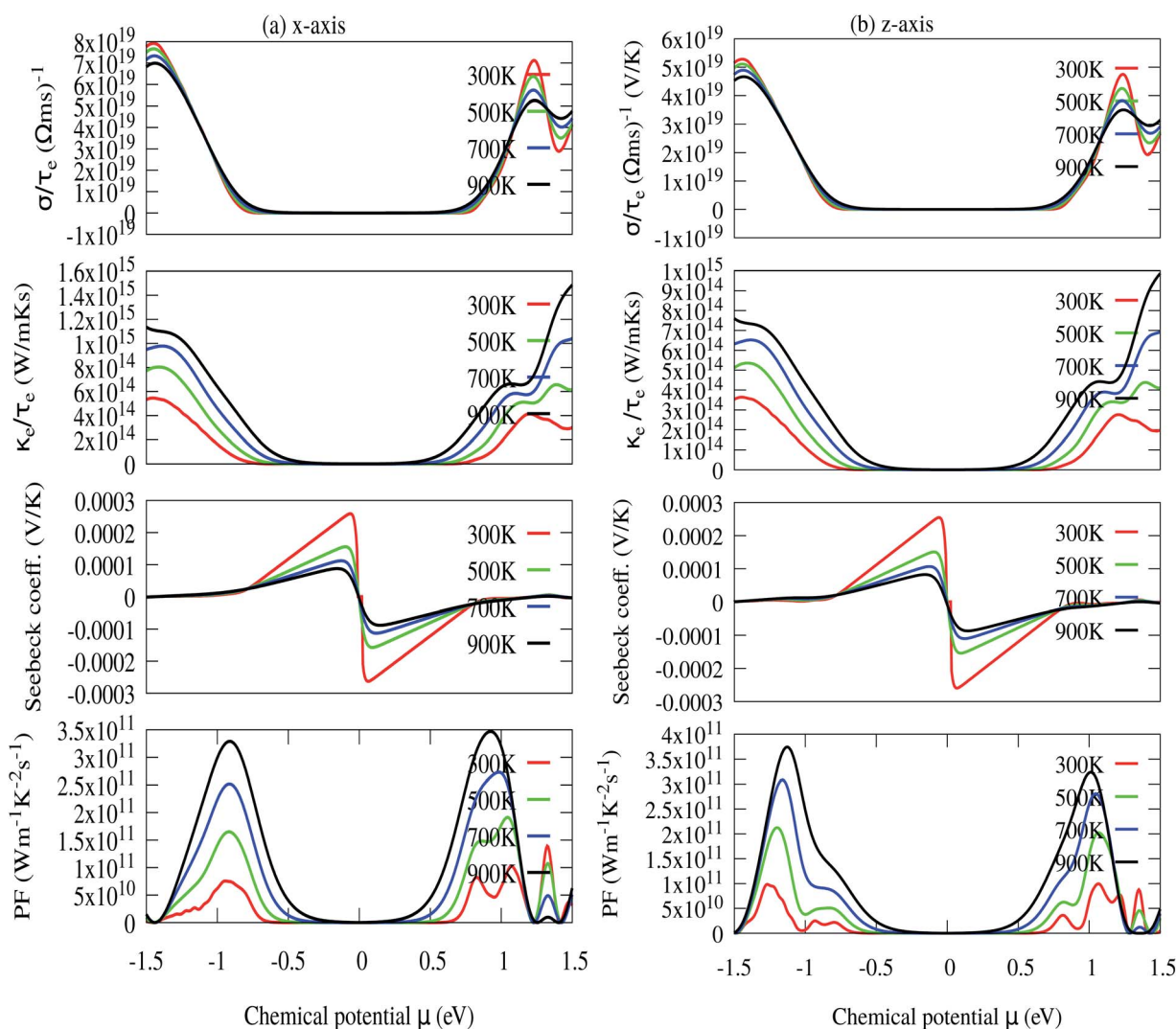


Fig. 7 (a) In-plane x -axis: electron relaxation time (τ_e) dependent electrical conductivity (σ/τ_e), electron thermal conductivity (κ_e/τ_e), Seebeck coefficient (S) and power factor (PF), (b) out-of-plane z -axis: electron relaxation time (τ_e) dependent electrical conductivity (σ/τ_e), electron thermal conductivity (κ_e/τ_e), Seebeck coefficient (S) and power factor (PF).



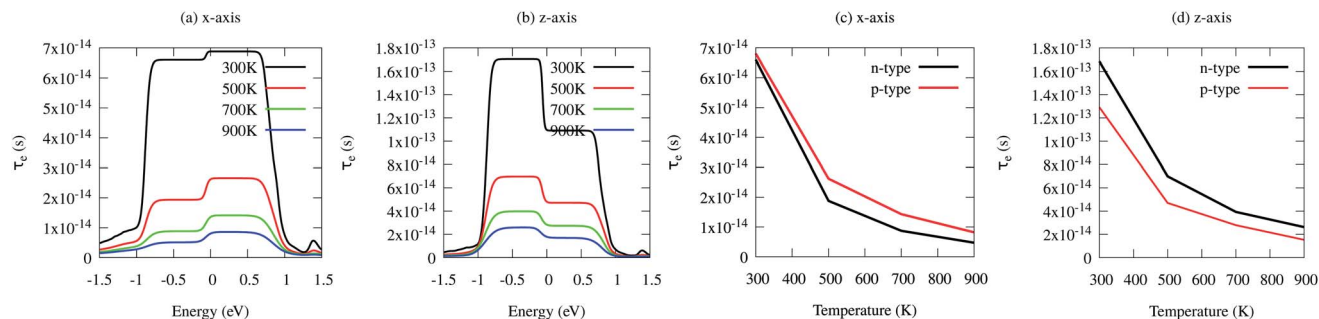


Fig. 8 (a) Electron relaxation time (τ_e) as a function of chemical potential along the x -axis, (b) electron relaxation time (τ_e) as a function of chemical potential along the z -axis, (c) electron relaxation time (τ_e) as a function of temperature along the x -axis and (d) electron relaxation time (τ_e) as a function of temperature along the z -axis.

energy. The τ_e value increases from $\sim 1 \times 10^{-14}$ s to $\sim 7 \times 10^{-14}$ s along the x -axis and from $\sim 2.0 \times 10^{-14}$ s to $\sim 1.8 \times 10^{-13}$ s along the z -axis as the temperature decreases from 900 K to 300 K [see Fig. 8(a–d)].

The optimized values of the thermoelectric parameters obtained at 300 K are presented in Table 2. The high value of S and low κ_1 are indicative that the monolayer MoS₂ system can be a prospective candidature for thermoelectric applications. However, the presence of a wide band gap in monolayer MoS₂ may result in a low value of electrical conductivity (σ). The use of SOC has a significant effect on the band energy of MoS₂ due to the sizeable spatial expansion of the Mo-3d-orbital which may lead to desirable physical and chemical properties. The spin-orbit interaction is applied along the easy spin quantization axis [001] direction. We found that the large GGA band gap of the monolayer MoS₂ semiconductor was reduced to ~ 1.68 eV which may enhance the power factor. Hence, we have calculated all the thermoelectric properties using GGA-SOC. The calculated σ for both n-type and p-type carriers along the x -axis are found to decrease from $\sim 5.0 \times 10^5$ to $\sim 5 \times 10^4 \Omega^{-1} \text{m}^{-1}$ with an increase in absolute temperature from 300 to 900 K. Similarly, along the z -axis, σ varies from $\sim 1.45 \times 10^6$ to $\sim 1.5 \times 10^5 \Omega^{-1} \text{m}^{-1}$ in the same temperature range. At room temperature the total thermal conductivities are found to be $\kappa_{xx} = \sim 36.23 \text{ W m}^{-1} \text{ K}^{-1}$ and $\kappa_{xx} = \sim 30.18 \text{ W m}^{-1} \text{ K}^{-1}$ for n-type and p-type carriers, respectively. On the other hand, the calculated room temperature values of the total thermal conductivities along the z -axis are almost two times higher, $\kappa_{zz} = \sim 60.00 \text{ W m}^{-1} \text{ K}^{-1}$ and $\kappa_{zz} = \sim 50.84 \text{ W m}^{-1} \text{ K}^{-1}$ for n-type

and p-type carriers, respectively. Our results for total thermal conductivity measured along the x -axis are consistent with the results obtained from molecular dynamics ($23.2 \text{ W m}^{-1} \text{ K}^{-1}$) and from a non-equilibrium Green's function ($26.2 \text{ W m}^{-1} \text{ K}^{-1}$).⁵² Moreover, the results along the z -axis agree well with the values of $52 \text{ W m}^{-1} \text{ K}^{-1}$ (ref. 35) and $85\text{--}110 \text{ W m}^{-1} \text{ K}^{-1}$ (ref. 54) measured for the vapour phase of few-layer MoS₂ and the (001) orientation of a MoS₂ crystal with basal plane thermal conductivity as a function of laser spot size, respectively. We observed a sharp decrease in κ as the temperature increased from 300 K to 900 K [Fig. 9(a and b)]. A similar trend has also been reported for WS₂, in which κ_1 decreases from 150 to 100 $\text{W m}^{-1} \text{ K}^{-1}$ within the 200–500 K temperature range. Also, in WSe₂, κ_1 decreases from 50 to 30 $\text{W m}^{-1} \text{ K}^{-1}$ in the same temperature range.⁸¹ The sharp rise in DOS near the Fermi energy gives rise to the high value of S . The maximum values of S are found to be $\sim 1.19 \times 10^{-4} \text{ V K}^{-1}$ and $\sim 1.41 \times 10^{-4} \text{ V K}^{-1}$ for the n-type and the p-type carriers along the x -axis, respectively. Fig. 9(a and b) display the total thermal conductivity ($\kappa = \kappa_1 + \kappa_e$) and figure of merit (ZT) as a function of chemical potential along the x - and z -axes, respectively. For both the x - and z -axes, the ZT value due to hole carriers (p-type) surpasses the n-type ZT value. The ZT values are found to be 0.60–0.76 at room temperature which seems to be too small for practical applications [Table 2]. Interestingly, the ZT increases up to ~ 1.00 at 1000 K for p-type carriers (taken from the peak value) [see Fig. 9(a and b)]. The linear behaviour of ZT as a function of temperature signifies the potential of monolayer MoS₂ as a high temperature thermoelectric material.

Table 2 Calculated σ ($\Omega^{-1} \text{m}^{-1}$), S (V K^{-1}), τ_e (s), $\kappa = \kappa_e + \kappa_1$ ($\text{W m}^{-1} \text{ K}^{-1}$) and thermoelectric efficiency (ZT) at 300 K (all calculations are performed with GGA-SOC)

Carriers	σ	S	τ_e	$\kappa = \kappa_e + \kappa_1$	ZT
Along xx					
n-type	5.13×10^5	-0.119×10^{-3}	6.807×10^{-14}	36.238	0.75
p-type	3.26×10^5	0.141×10^{-3}	6.602×10^{-14}	30.180	0.76
Along zz					
n-type	6.192×10^5	-1.12×10^{-4}	1.648×10^{-13}	60.00	0.60
p-type	14.07×10^5	1.25×10^{-4}	1.291×10^{-13}	50.84	0.62



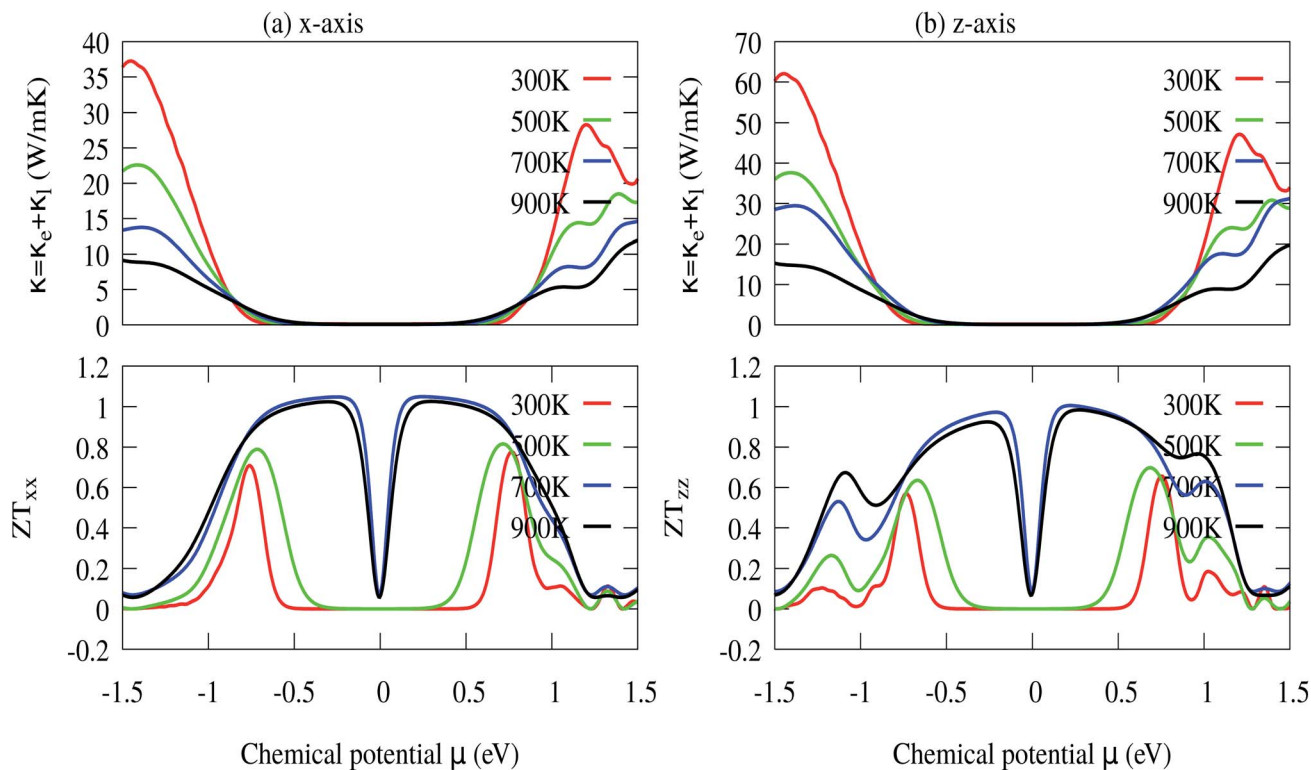


Fig. 9 (a) In-plane x-axis: total thermal conductivity ($\kappa = \kappa_e + \kappa_l$) and ZT_{xx} , and (b) out-of-plane z-axis: total thermal conductivity ($\kappa = \kappa_e + \kappa_l$) and ZT_{zz} .

4 Conclusion

In this investigation, we have studied the electronic and thermoelectric properties of monolayer MoS_2 using the GGA and GGA-SOC approaches. The calculation of the electronic structure shows that monolayer MoS_2 is a direct band gap semiconductor, as the CBM and VBM lie at the same K-symmetry point. The calculated band gap is found to be 1.75 eV, in good agreement with the previous experimental and theoretical results. Monolayer MoS_2 exhibits high S and low κ values, which are crucial for thermoelectric applications. However, with GGA the electrical conductivity (σ) is observed to be suppressed which eventually limits the thermoelectric power factor due to the presence of a wide band gap. We have found that SOC has a significant effect on the band energy of monolayer MoS_2 due to the presence of the larger 4d orbital. A reduction of the direct band gap has been observed on application of spin-orbit coupling along the spin quantization 001 direction. The reduced band gap is expected to enhance σ at room temperature. Our calculated thermoelectric parameters are consistent with the available data. The reduced lattice thermal conductivity at elevated temperatures is another interesting feature. As a result, the ZT value approaches the benchmark value of ~ 1.0 at a temperature of ~ 1000 K.

Conflicts of interest

All authors declare that there are no conflicts of interest.

Acknowledgements

D. P. Rai acknowledges a Core Research Grant from the Department of Science and Technology SERB (CRG DST-SERB, Govt. of India, New Delhi) via Sanction no. CRG/2018/000009(Ver-1). A. Laref acknowledge the Research Center of Female Scientific and Medical College, King Saud University for financial support. Sohail Ahmad would like to express gratitude to King Khalid University, Abha, Saudi Arabia for administrative and technical support. Tuan V. Vu acknowledges the Vietnam National Foundation for Science and Technology Development (NAFOSTED) under grant number 103.01-2018.334.

Notes and references

- 1 F. Schwierz, Graphene Transistors: Status, Prospects, and Problems, *Proc. IEEE*, 2013, **101**, 1567–1584.
- 2 X. Huang, Z. Y. Yin, S. X. Wu, X. Y. Qi, Q. Y. He, Q. C. Zhang, Q. Y. Yan, F. Boey and H. Zhang, Graphene-based materials: synthesis, characterization, properties, and applications, *Small*, 2011, **7**, 1876–1902.
- 3 D. P. Rai, S. Kaur and S. Srivastava, Band gap modulation of mono and bi-layer hexagonal ZnS under transverse electric field and bi-axial strain: a first principles study, *Phys. B*, 2018, **531**, 90–94.
- 4 Y. Kim, J. L. Huang and C. M. Lieber, Characterization of nanometer scale wear and oxidation of transition metal



- dichalcogenide lubricants by atomic force microscopy, *Appl. Phys. Lett.*, 1991, **59**, 3404–3406.
- 5 K. H. Hu, X. G. Huand and X. J. Sun, Morphological effect of MoS₂ nanoparticles on catalytic oxidation and vacuum lubrication, *Appl. Surf. Sci.*, 2010, **256**, 2517–2523.
 - 6 E. Fortin and W. Sears, Photovoltaic effect and optical absorption in MoS₂, *J. Phys. Chem. Solids*, 1982, **43**, 881–884.
 - 7 A. H. Reshak and S. Auluck, Calculated optical properties of 2H-MoS₂ intercalated with lithium, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **68**, 125101–125107.
 - 8 K. K. Liu, W. Zhang, Y. H. Lee, Y. C. Lin, M. T. Chang, C. Y. Su, C. S. Chang, H. Li, Y. Shi and H. Zhang, Growth of Large-Area and Highly Crystalline MoS₂ Thin Layers on Insulating Substrates, *Nano Lett.*, 2012, **12**, 1538–1544.
 - 9 Y. H. Lee, X. Q. Zhang, W. Zhang, M. T. Chang, C. T. Lin, K. D. Chang, Y. C. Yu, J. T. W. Wang, C. S. Chang, L. J. Li and T. W. Lin, Synthesis of large-area MoS₂ atomic layers with chemical vapor deposition, *Adv. Mater.*, 2012, **24**, 2320–2325.
 - 10 R. J. Smith, P. J. King, M. Lotya, C. Wirtz, U. Khan, S. De, A. O'Neill, G. S. Duesberg, J. C. Grunlan and G. Moriarty, Large Scale Exfoliation of Inorganic Layered Compounds in Aqueous Surfactant Solutions, *Adv. Mater.*, 2011, **23**, 3944–3948.
 - 11 C. Lee, H. Yan, L. E. Brus, T. F. Heinz, J. Hone and S. Ryu, Anomalous Lattice Vibrations of Single- and Few-Layer MoS₂, *ACS Nano*, 2010, **4**, 2695–2700.
 - 12 B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti and A. Kis, Single-layer MoS₂ transistors, *Nat. Nanotechnol.*, 2011, **6**, 147–150.
 - 13 R. Yan, J. R. Simpson, S. Bertolazzi, J. Brivio, M. Watson, X. Wu, A. Kis, T. Luo, A. R. H. Walker and H. G. Xing, Thermal Conductivity of Monolayer Molybdenum Disulfide Obtained from Temperature-Dependent Raman Spectroscopy, *ACS Nano*, 2014, **8**, 986–993.
 - 14 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, Atomically Thin MoS₂: A New Direct-Gap Semiconductor, *Phys. Rev. Lett.*, 2010, **105**, 136805.
 - 15 D. P. Rai, V. V. Tuan, A. Laref, M. P. Ghimire, P. K. Patra and S. Srivastava, Electronic and optical properties of 2D monolayer (ML) MoS₂ with vacancy defect at S sites, *Nano-Struct. Nano-Objects*, 2020, **21**, 100404, DOI: 10.1016/j.nanoso.2019.100404.
 - 16 F. J. Urbanos, A. Black, R. B. Gavito, A. L. Vázquez de Parga, R. Miranda and D. Granados, Electrical and geometrical tuning of MoS₂ field effect transistors via direct nanopatterning, *Nanoscale*, 2019, **11**, 11152–11158.
 - 17 T. Pham, G. Li, E. Bekyarova, M. E. Itkis and A. Mulchandani, MoS₂-based optoelectronic gas sensor with sub-parts-per-billion limit of NO₂ gas detection, *ACS Nano*, 2019, **13**, 3196–3205.
 - 18 N. Goel, R. Kumar and M. Kumar, Enhanced sensing response with complete recovery of MoS₂ sensor under photoexcitation, *AIP Conf. Proc.*, 2018, **1942**, 050060–050064.
 - 19 L. Zhe, M. Nasir, T. Muhammad, P. Lun, Z. Xiangwen and Z. Ji-Jun, Fabrication of zero to three dimensional nanostructured molybdenum sulfides and their electrochemical and photocatalytic applications, *Nanoscale*, 2016, **8**, 18250–18269.
 - 20 P. Zhao, A. Khosravi, A. Azcatl, P. Bolshakov, G. Mirabelli, E. Caruso, C. L. Hinkle, P. K. Hurley, R. M. Wallace and C. D. Young, Evaluation of border traps and interface traps in HfO₂/MoS₂ gate stacks by capacitance-voltage analysis, *2D Mater.*, 2018, **5**, 031002–031008.
 - 21 X. He, H. Li, Z. Zhu, Z. Dai, Y. Yang, P. Yang, Q. Zhang, P. Li, U. Schwingenschlögl and X. Zhang, Strain engineering in monolayer WS₂, MoS₂, and the WS₂/MoS₂ heterostructure, *Appl. Phys. Lett.*, 2016, **109**, 173105.
 - 22 H. V. Phuc, N. N. Hieu, B. D. Hoi, N. V. Hieu, T. V. Thu, N. M. Hung, V. V. Ilyasov, N. A. Poklonski and C. V. Nguyen, Tuning the Electronic Properties, Effective Mass and Carrier Mobility of MoS₂ Monolayer by Strain Engineering: First-Principle Calculations, *J. Electron. Mater.*, 2018, **47**, 730–736.
 - 23 A. Kumar and P. Ahluwalia, A first principle comparative study of electronic and optical properties of 1H-MoS₂ and 2H-MoS₂, *Mater. Chem. Phys.*, 2012, **135**, 755–761.
 - 24 K. P. Dhakal, D. L. Duong, J. Lee, H. Nam, M. Kim, M. Kan, Y. H. Lee and J. Kim, Confocal absorption spectral imaging of MoS₂: optical transitions depending on the atomic thickness of intrinsic and chemically doped MoS₂, *Nanoscale*, 2014, **6**, 13028–13035.
 - 25 N. N. Hieu, V. V. Ilyasov, T. V. Vu, N. A. Poklonski, H. V. Phuc, L. T. T. Phuong, B. D. Hoi and C. V. Nguyen, First principles study of optical properties of molybdenum disulfide: from bulk to monolayer, *Superlattices Microstruct.*, 2018, **115**, 10–18.
 - 26 Q. H. Wang, K. K. Zadeh, A. Kis, J. N. Coleman and M. S. Strano, Electronics and optoelectronics of two-dimensional transition metal dichalcogenides, *Nat. Nanotechnol.*, 2012, **7**, 699–712.
 - 27 S. Sharma, S. Kumar and U. Schwingenschlögl, Arsenene and Antimonene: Two-Dimensional Materials with High Thermoelectric Figures of Merit, *Phys. Rev. Appl.*, 2017, **8**, 044013–044018.
 - 28 M. Tahir and U. Schwingenschlögl, Tunable thermoelectricity in monolayers of MoS₂ and other group-VI dichalcogenides, *New J. Phys.*, 2014, **16**, 115003–115011.
 - 29 G. H. Hong, Y. Teng, T. Peng and Z. Z. Dong, Theoretical study of thermoelectric properties of MoS₂, *Chin. Phys. B*, 2014, **23**, 017201–017207.
 - 30 D. P. Rai, A. Shankar, Sandeep, M. P. Ghimire, R. Khenata and R. K. Thapa, Study of the enhanced electronic and thermoelectric (TE) properties of Zr_xHf_{1-x-y}Ta_yNiSn: a first principles study, *RSC Adv.*, 2015, **5**, 95353–95359.
 - 31 K. Kaur, R. Kumar and D. P. Rai, A promising thermoelectric response of HfRhSb half Heusler compound at high temperature: a first principle study, *J. Alloys Compd.*, 2018, **763**, 1018–1023.
 - 32 K. Kaur, D. P. Rai, R. K. Thapa and S. Srivastava, Structural, electronic, mechanical, and thermoelectric properties of a novel half Heusler compound HfPtPb, *J. Appl. Phys.*, 2017, **122**, 045110–045117.



- 33 D. P. Rai, Sandeep, A. Shankar, R. Khenata, A. H. Reshak, C. E. Ekuma, R. K. Thapa and S.-H. Ke, Electronic, optical, and thermoelectric properties of $\text{Fe}_{2+x}\text{V}_{1-x}\text{Al}$, *AIP Adv.*, 2017, **7**, 045118.
- 34 D. Wickramaratne, F. Zahid and R. K. Lake, Electronic and thermoelectric properties of few-layer transition metal dichalcogenides, *J. Chem. Phys.*, 2014, **140**, 124710–124713.
- 35 S. Sahoo, A. P. S. Gaur, M. Ahmadi, M. J. F. Guinel and R. S. Katiyar, Temperature-Dependent Raman Studies and Thermal Conductivity of Few-Layer MoS_2 , *J. Phys. Chem. C*, 2013, **117**, 9042–9047.
- 36 X. Wei, Y. Wang, Y. Shen, G. Xie, H. Xiao, J. Zhong and G. Zhang, Phonon thermal conductivity of monolayer MoS_2 : a comparison with single layer graphene, *Appl. Phys. Lett.*, 2014, **105**, 103902–103905.
- 37 G. Zhang and Y. W. Zhang, Thermoelectric properties of two-dimensional transition metal dichalcogenides, *J. Mater. Chem. C*, 2017, **5**, 7684–7698.
- 38 D. Qin, P. Yan, G. Ding, X. Ge and G. Gao, Monolayer PdSe_2 : a promising two-dimensional thermoelectric material, *Sci. Rep.*, 2018, **8**, 2764–2768.
- 39 K. Hippalgaonkar, Y. Wang, Y. Ye, H. Zhu, Y. Wang, J. Moore and X. Zhang, High thermoelectric power factor in two-dimensional crystals of MoS_2 , *Phys. Rev. B*, 2017, **95**, 115407–115409.
- 40 Z. Jin, Q. Liao, H. Fang, Z. Liu, W. Liu, Z. Ding, T. Luo and N. Yang, A Revisit to High Thermoelectric Performance of Single-layer MoS_2 , *Sci. Rep.*, 2015, **5**, 18342–18347.
- 41 W. Huang, H. Da and G. Liang, Thermoelectric performance of MX_2 ($\text{M} = \text{Mo}, \text{W}$; $\text{X} = \text{S}, \text{Se}$) monolayers, *J. Appl. Phys.*, 2013, **113**, 104304–104307.
- 42 S. Kumar and U. Schwingenschlöggl, Thermoelectric Response of Bulk and Monolayer MoSe_2 and WSe_2 , *Chem. Mater.*, 2015, **27**, 1278–1284.
- 43 A. N. Gandhi and U. Schwingenschlöggl, WS_2 As an Excellent High-Temperature Thermoelectric Material, *Chem. Mater.*, 2014, **26**, 6628–6637.
- 44 Y. Su, M. A. Ebrish, E. J. Olson and S. J. Koester, SnSe_2 field-effect transistors with high drive current, *Appl. Phys. Lett.*, 2013, **103**, 263104.
- 45 A. Arab, A. V. Davydov, D. A. Papaconstantopoulos and Q. Li, Monolayer MoS_2 Nanoribbons as a Promising Material for Both n-type and p-type Legs in Thermoelectric Generators, *J. Electron. Mater.*, 2016, **45**, 5253–5263.
- 46 H. Lv, W. Lu, D. Shao, H. Lu and Y. Sun, Strain-induced enhancement in the thermoelectric performance of a ZrS_2 monolayer, *J. Mater. Chem. C*, 2016, **4**, 4538–4545.
- 47 G. Ding, G. Y. Gao, Z. Huang, W. Zhang and K. Yao, Thermoelectric properties of monolayer MSe_2 ($\text{M} = \text{Zr}, \text{Hf}$): low lattice thermal conductivity and a promising figure of merit, *Nanotechnology*, 2016, **27**, 375703–375707.
- 48 C. Chiritescu, D. G. Cahill, N. Nguyen, D. Johnson, A. Bodapati, P. Koblinski and P. Zchack, Ultralow Thermal Conductivity in Disordered, Layered WSe_2 Crystals, *Science*, 2007, **315**, 351–353.
- 49 V. Varshney, S. S. Patnaik, C. Muratore, A. K. Roy, A. A. Voevodin and B. L. Farmer, MD simulations of molybdenum disulphide (MoS_2): force-field parameterization and thermal transport behavior, *Comput. Mater. Sci.*, 2010, **48**, 101–108.
- 50 J. W. Jiang, H. S. Park and T. Rabezuk, Molecular dynamics simulations of single-layer molybdenum disulphide (MoS_2): Stillinger-Weber parametrization, mechanical properties, and thermal conductivity, *J. Appl. Phys.*, 2013, **114**, 064307–064310.
- 51 X. J. Liu, G. Zhang, Q. X. Pei and Y. W. Zhang, Phonon thermal conductivity of monolayer MoS_2 sheet and nanoribbons, *Appl. Phys. Lett.*, 2013, **103**, 133113–133115.
- 52 Y. Cai, J. Lan, G. Zhang and Y. W. Zhang, Lattice vibrational modes and phonon thermal conductivity of monolayer MoS_2 , *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**, 035438.
- 53 W. Li, J. Carrete and N. Mingo, Thermal conductivity and phonon linewidths of monolayer MoS_2 from first principles, *Appl. Phys. Lett.*, 2013, **103**, 253103–253104.
- 54 J. Liu, G. M. Choi and D. G. Cahill, Measurement of the anisotropic thermal conductivity of molybdenum disulfide by the time-resolved magneto-optic Kerr effect, *J. Appl. Phys.*, 2014, **116**, 233107.
- 55 K. Kaur, D. Murali and B. R. K. Nanda, Stretchable and dynamically stable promising two-dimensional thermoelectric materials: ScP and ScAs , *J. Mater. Chem. A*, 2019, **7**, 12604–12615.
- 56 M. Buscema, M. Barkelid, V. Zwiller, H. S. van der Zant and G. A. Steele, Large and Tunable Photothermoelectric Effect in Single-Layer MoS_2 , *Nano Lett.*, 2013, **13**, 358–363.
- 57 G. K. H. Madsen and D. J. Singh, BoltzTraP. A code for calculating band-structure dependent quantities, *Comput. Phys. Commun.*, 2006, **175**, 67–71.
- 58 P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnicka, J. Luitz and K. Schwarz, *An Augmented Plane Wave plus Local Orbitals Program for Calculating Crystal Properties. Wien2K User's Guide*, Techn. Universitat Austria, Wien, 2008.
- 59 P. Giannozzi, *et al.*, QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials, *J. Phys.: Condens. Matter*, 2009, **21**, 395502–395519.
- 60 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 61 S. Ahmad and S. Mukherjee, A Comparative Study of Electronic Properties of Bulk MoS_2 and Its Monolayer Using DFT Technique: Application of Mechanical Strain on MoS_2 Monolayer, *Graphene*, 2014, **3**, 52–59.
- 62 C. E. Ekuma, S. Najmaei and M. Dubey, Electronic and vibrational properties of van der Waals heterostructures of vertically stacked few-layer atomically thin MoS_2 and BP, *Mater. Today Commun.*, 2019, **19**, 383–392.
- 63 A. Togo, L. Chaput and I. Tanaka, Distributions of phonon lifetimes in Brillouin zones, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 094306–094331.
- 64 P. Johari and V. B. Shenoy, Tuning the Electronic Properties of Semiconducting Transition Metal Dichalcogenides by Applying Mechanical Strains, *ACS Nano*, 2012, **6**, 5449–5456.



- 65 Y. Ding, Y. Wang, J. Ni, L. R. Shi, S. Shi and W. Tang, First principles study of structural, vibrational and electronic properties of graphene-like MX_2 ($M = \text{Mo, Nb, W, Ta}$; $X = \text{S, Se, Te}$) monolayers, *Phys. B*, 2011, **406**, 2254–2260.
- 66 A. Splendiani, *et al.*, Emerging photoluminescence in monolayer MoS_2 , *Nano Lett.*, 2010, **10**, 1271–1275.
- 67 A. Kuc, N. Zibouche and T. Heine, Influence of quantum confinement on the electronic structure of the transition metal sulfide TS_2 , *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **83**, 245213–245214.
- 68 Y. U. Zhiyi, Y. X. Huang and S. C. Shen, Spin-orbit splitting of the valence bands in silicon determined by means of high-resolution photoconductive spectroscopy, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1989, **39**, 6287.
- 69 L. Chaput, A. Togo, I. Tanaka and G. Hug, Phonon–phonon interactions in transition metals, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 094302–094306.
- 70 X. Gonze and C. Lee, Dynamical matrices, Born effective charges, dielectric permittivity tensors, and interatomic force constants from density-functional perturbation theory, *Phys. Rev. B: Solid State*, 1997, **55**, 10355–10368.
- 71 S. Sharma, N. Singh and U. Schwingenschlöggl, Two-Dimensional Tellurene as Excellent Thermoelectric Material, *ACS Appl. Energy Mater.*, 2018, **1**, 1950–1954.
- 72 Y. Wang, Z. Lu and X. Ruan, First principles calculation of lattice thermal conductivity of metals considering phonon–phonon and phonon–electron scattering, *J. Appl. Phys.*, 2016, **119**, 225109–225110.
- 73 K. Park, A. Mohamed, M. Dutta, M. A. Strosio and C. Bayram, Electron Scattering *via* Interface Optical Phonons with High Group Velocity in Wurtzite GaN-based Quantum Well Heterostructure, *Sci. Rep.*, 2018, **8**, 15947.
- 74 S. Mukhopadhyay, L. Lindsay and D. J. Singh, Optic phonons and anisotropic thermal conductivity in hexagonal $\text{Ge}_2\text{Sb}_2\text{Te}_5$, *Sci. Rep.*, 2016, **6**, 37076–37078.
- 75 W. J. Liu, M. L. Liu, B. Liu, R. G. Quhe, M. Lei, S. B. Fang, H. Teng and Z. Y. Wei, Nonlinear optical properties of MoS_2 – WS_2 heterostructure in fiber lasers, *Opt. Express*, 2019, **27**, 6689–6699.
- 76 A. Kormányos, V. Zólyomi, N. D. Drummond and G. Burkard, Spin–Orbit Coupling, Quantum Dots, and Qubits in Monolayer Transition Metal Dichalcogenides, *Phys. Rev. X*, 2014, **4**, 011034.
- 77 J. Xi, T. Zhao, D. Wang and Z. Shuai, Tunable Electronic Properties of Two-Dimensional Transition Metal Dichalcogenide Alloys: A First-Principles Prediction, *J. Phys. Chem. Lett.*, 2014, **5**, 285–291.
- 78 F. Zahid, L. Liu, Y. Zhu, J. Wang and H. Guo, A generic tight-binding model for monolayer, bilayer and bulk MoS_2 , *AIP Adv.*, 2013, **3**, 052111–052116.
- 79 G. J. Snyder and E. S. Toberer, Complex thermoelectric materials, *Nat. Mater.*, 2008, **7**, 105–114.
- 80 M. Cutler, J. F. Leavy and R. L. Fitzpatrick, Electronic Transport in Semimetallic Cerium Sulfide, *Phys. Rev.*, 1964, **133**, A1143–A1152.
- 81 A. Mobaraki, A. Kandemir, H. Yapicioglu, O. G. Aijlseren and C. Sevik, Validation of inter-atomic potential for WS_2 and WSe_2 crystals through assessment of thermal transport properties, *Comput. Mater. Sci.*, 2018, **144**, 92–98.
- 82 G. Samsonidze and B. Kozinsky, Thermoelectric Materials: Accelerated Screening of Thermoelectric Materials by First-Principles Computations of Electron–Phonon Scattering, *Adv. Energy Mater.*, 2018, **8**(20), 1800246.
- 83 S. Bang, J. Kim, D. Wee, G. Samsonidze and B. Kozinsky, Estimation of electron–phonon coupling *via* moving least squares averaging: a method for fast-screening potential thermoelectric materials, *Materials Today Physics*, 2018, **6**, 22–30.

