



Unravelling the true MOF-5 luminescence†

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Highly pure millimeter-sized MOF-5 single crystals were synthesized and characterized. Photoluminescence (PL) spectroscopy and time-correlated single photon counting (TCSPC) demonstrate a solvent-guest dependency of MOF-5 emission and its ligand-centred nature. These results allow measuring the true MOF-5 luminescence free of solvent at a wavelength of 355 nm, a significantly lower wavelength than previously published. MOF-5 emission was also evaluated with different solvents and various degrees of water intake, explaining previously published observations. Comparison between lifetimes shows the fluorophore stabilization within the frameworks and demonstrates the progressive influence of the Zn₄O subunits on the fluorescence during hydration. Overall, this work highlights the necessity to obtain phase-pure material, especially when moisture sensitivity can play a role, before ascribing electronic transitions. This study is a rigorous new take on the iconic MOF-5 and on its photoluminescence properties.

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Metal–organic frameworks (MOFs) are crystalline hybrid materials, composed of metal clusters that are linked by organic ligands. Due to their unique topology, these materials offer promising applications such as separation, catalysis, drug delivery or gas storage among others.¹ While many contributions have demonstrated the use of MOFs as luminescent sensors their photophysical properties still draw attention after many years.^{2–6} Indeed, due to their hybrid nature, photoluminescence (PL) can arise from various sites: metallic nodes, organic linkers or interaction between both of them such as MLCT, LMCT, LLCT and MMCT (M: metal, L: ligand, and CT: charge transfer). Due to their near thermodynamic density of defect,⁷ the luminescence of these are negligible in crystalline phase. Furthermore, molecules trapped in the porosity can also emit, influence or induce luminescence, giving an additional complexity to photophysical comprehension.⁸ The present work demonstrates the influence of frameworks (especially the one based on Zn₄O subunits) on linker PL and attests the necessity to obtain pure phase material to conclude on mechanisms leading to MOF luminescence. The well know MOF-5 (: IRMOF-1) was the subject of numerous studies during the past two decades.^{9–12} However, to the best of our knowledge there is no report on a ligand centered (LC) emission. Moreover, there is still no consensus on MOF-5 emission, with reported fluorescence between 390 nm^{11,12} and 525 nm^{9,10} and different assigned transitions. Thus, we chose MOF-5 as a case study to explain

how coherently interpret photophysical properties of MOFs based on Zn₄O subunits, and conclude on the luminescence of this iconic MOF. MOF-5 luminescence was first studied by Bordiga *et al.*, with a reported maximum of emission centred at 525 nm. They assigned this emission as a charge transfer from benzene dicarboxylic acid to the Zn₄O₁₃ metal clusters.⁹ Later, Tachikawa *et al.* studied MOF-5 nanoparticles luminescence and found that the responsible transition for the green emission was similar to that of ZnO quantum dots (QDs). Therefore, they attributed the emission to MOF-trapped ZnO nanoparticles instead of MOF.¹⁰ Allendorf and co-workers were the first team to recognize the discrepancy between published results and underlined the presence of impurities within the frameworks. They synthesized MOF-5 with different processes, generating more or less pure materials and studied their PL properties. They were able to reproduce previous results from Bordiga and Tachikawa on frameworks including ZnO impurities. Above all, they highlighted the misunderstanding of previous reports which interpret emissions from impurities due to poor preparation. MOF-5 emission was then ascribed to a transition occurring at 397 nm upon excitation at 345 nm.¹¹ A 395 nm emission would imply a strong electronic interaction between the benzene-1,4-dicarboxylic acid (H₂BDC) ligand and the Zn cluster. Recent DFT studies compare geometry and IR spectra between electronically excited state and ground state of MOF-5.¹² These investigations reveal the rigidity of Zn₄O₁₃ clusters compared to non-rigid BDC²⁻ ligands which leads to similar calculated emission band for MOF-5 and free H₂BDC. Results were quite similar with Allendorf's work and their interpretation suggests that the luminescence was carried out by LLCT mechanism with little contribution of the metallic cluster.

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Motivated by both experimental and theoretical reports lacking of consistency,⁶ the following work goal is to reproduce, explain and unify previous results. Hence this study focuses on obtaining highly pure phase material before studying PL properties and to ascribe a mechanism to MOF-5 emission. We were thus able to observe a UV emission from MOF-5 single crystals, and the PL dependence with solvents and water molecules which slowly diffuse within the crystal and collapse its structure.

As discussed above, poor preparation may be responsible for misunderstanding of PL properties requiring the synthesis of pure phase material.¹¹ A focus was made in our laboratory to obtain large millimeter-sized MOF-5 single crystals in a reproducible manner with high purity. They were synthesized (see ESI†) from zinc nitrate hexahydrate ($\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$) and benzene-1,4-dicarboxylic acid (H_2BDC) in *N,N*-diethylformamide (DEF) using a solvothermal procedure which was slightly different from the one reported by Yaghi *et al.*^{13,14} Colorless, large single crystals were immediately preserved from moisture inside a N_2 glovebox and washed with anhydrous solvents to avoid any framework decomposition as we will see below. The purity was assessed by several structural characterizations: X-ray diffraction, thermal and infrared analyses (ESI†). Fig. 1 represents powder X-ray diffraction (PXRD) data. Refinement on single crystal X-ray data permits to calculate the associated PXRD pattern (black) that is in perfect agreement with previous reports on MOF-5.^{15,16} Following activation, thermogravimetric analysis (TGA) demonstrated characteristic thermal decomposition of the frameworks with three distinct regions assigned to H_2O /solvent desorption and structural decomposition of the framework.¹⁶ More importantly, the weight loss of our sample upon decomposition into ZnO is 58% which perfectly fits the theoretical calculated weight loss of original MOF-5, 57.7% (ESI, Fig. S1†) indicating that there is no remaining ZnO or other guest impurities in our crystals. Raman and BET analysis were also performed and show results in

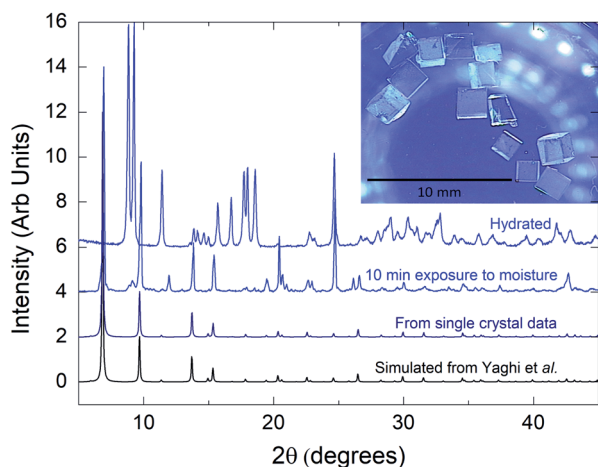


Fig. 1 Powder X-ray diffraction data. Intensities are normalized to the maximum. The pattern for MOF-5 was calculated from single crystal structural data (see ESI†). Inset is a picture of single crystals.

complete agreement with previous reports (ESI, Fig. S2 and S3†).

Photoluminescence investigations were performed (Fig. 2) on a pure single crystal soaked in different solvents and compared to the H_2BDC building unit dissolved in these same solvents. Reasons to collect PL data on MOF full of solvent was motivated by the observations of strong solvent effect on the H_2BDC fluorescence. Results show indeed different maximum wavelengths for MOF-5 from one solvent to another. When impregnated with DMF or DMSO, the framework exhibits a maximum of emission at 342 or 363 nm, respectively. The smaller solvatochromic shift observed in DMF comes with a reduced broadening of the band that is observed in DMF for the free ligand and the MOF.²⁻¹⁷ MOF emission shows a 5 nm bathochromic shift compared to the ligand for DMF. These observations are substantially the same for MOF-5 in DMSO, which has the same spectrum profile than H_2BDC , with the difference that framework emission presents a hypsochromic shift of 4 nm. It is well known that the dipole moment of the

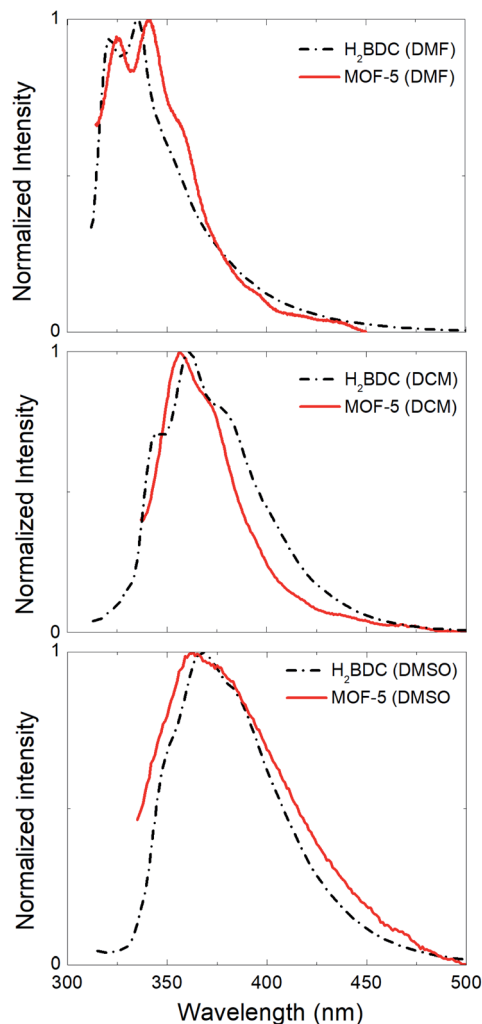


Fig. 2 Steady-state emission spectra for H_2BDC (dashed) and MOF-5 (solid), respectively dissolved and soaked, in DMF (top), DCM/ CH_2Cl_2 (middle) and DMSO (bottom) under 300 nm excitation.



solvent, size of the solute molecules and their interaction with the solvent play a major role on the system's global fluorescence. The solvatochromism is due to the difference in the solvation of the excited state with respect to the ground state. It has been described as a stabilization of the electric dipole of both states with the dielectric constant and refractive index of the solvent.¹⁸ This phenomenon is also visible for highly pure single crystals. The impact of the solvent on MOF-5 structure is negligible¹⁹ compared to other breathing MOF, we discarded this hypothesis as a reason of the solvatochromism. Our interpretation is that the solvent behaviour within the framework is similar to its bulk. Solvent molecules can enter into the pores and form a solvation environment similar to a solution one. One can note here that we are treating the absorbed molecules as a solvation shell from the point of view of the BDC linker. Chemi- and physi-sorption interaction are similar in both cases, balancing hydrogen bonding and van der Waals forces. In case of DMF, DMSO and later water, we think H-bond between the MOF surface and the molecules, as well as between molecules themselves, are the predominant factor of molecules immobilisation inside a pore. As shown in Fig. 2, the similarity in wavelengths and vibronic structure points towards luminescence in MOF-5 arising from a single excited state S_1 which is similar to the H_2BDC in solution and thus, suggests a ligand-centred emission only. Decay time measurements allow us to confirm our hypothesis, revealing a monoexponential decay, characteristic of a single photophysical process, and the same lifetime for frameworks and free linker in both solvents (ESI, Fig. S4 and S5†).

To remove the effect of the solvent and investigate only MOF-5 fluorescence, solvent molecules were removed from porosity with extreme care as the framework can collapse due to capillary forces while activating.²⁰ The resulting PLE and PL spectra (black) are shown in Fig. 3, revealing excitation and emission maxima at 313 nm and 358 nm, respectively. The symmetry of the excitation and emission coupled with vibronic structure suggest that the transition occurs from a single excited state S_1 to the fundamental state S_0 as seen above. However, and compared to Fig. 2, we can notice a broadening of the emission

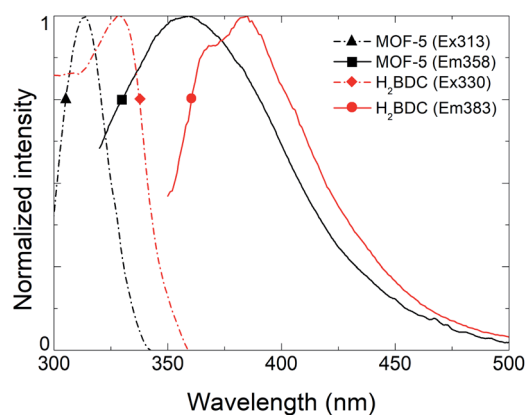


Fig. 3 Comparison of excitation (dashed) and emission (solid) spectra for an activated MOF-5 (black) and solid H_2BDC (blue).

which we attribute to a partial collapse of the framework. We assume that the shift to shorter wavelengths of both emission and excitation maxima compared to H_2BDC are the consequence of an increase of the band gap due to stabilization of linker within the framework and the absence of adsorbed molecules. Indeed, the decrease of the Stokes shift confirms this idea as the thermal states are getting closer to the fundamental and excited states (ESI, Fig. S7†). Moreover, a recent report²¹ on MOF-5 study by DFT shows the electronic band structure of cubic frameworks based on Zn_4O units and finds that its band gap should be located in the 3.56–3.46 eV range, corresponding to an emission between 348 nm and 358 nm, which is consistent with our observation. Time-correlated Single Photon Counting (TCSPC) measurements were performed on solid sample and demonstrate that evacuated MOF-5 lifetime at 355 nm is comparable with benzenecarboxylic disodic salt (Na_2BDC) with a heavier fast components. This fast component is closer to the solubilized H_2BDC than the solid state confirming the idea that the framework stabilizes the fluorophore in the same way a solvation shell would. (ESI, Fig. S8†).

The above observation contrasts with previous reports on MOF-5 photoluminescence. However, this is not surprising regarding the architecture of this type of framework. The secondary building unit (SBU), composed of Zn^{II} is, as well as Cd^{II} , favorable to have a luminescence carried out by linkers only. The reason resides in the d^{10} configuration of these ions. The absence of potential d–d transitions allows only luminescence from ligand.²² The SBU is not spectator so far as it varies the length of the ligand and so on the distribution of π -electron responsible of fluorescence.

As luminescence is carried out by the ligand only and dependent on the surroundings, we assume that any degradation affecting the structure will impact the associated PL spectra. As already observed in many contributions, MOF-5 is moisture sensitive.^{13,23,24} In Fig. 1, only 10 min moisture exposure results in the appearance of a new crystalline phase and longer exposure results in the appearance of a new emitting species which was never ascribed but nonetheless already observed.¹⁶ Allendorf and co-workers were one of the first to highlight the presence of a 445 nm emitting species.¹¹ Despite no explanation, this phenomenon can be explained by water adsorption and insertion into the frameworks, involving the Zn_4O SBU in the fluorescence process. As studied by Siegel and co-worker water adsorption and insertion in MOF-5 behaves in two steps. First stage consists of adsorption of H_2O molecules close to $Zn-O$ clusters. Then the second stage consists in water insertion *via* breaking of $Zn-O$ bonds²³ as shown in Fig. 4a. Fig. 4b shows different PL spectra of a MOF-5 crystal soaked in DMF and exposed to moisture. We can observe a rapid shift of the framework trident shaped emission from 342 nm to 365 nm and the apparition of a broad emission band centred at 445 nm. Then the crystal was evacuated from DMF and left several days in open air, leading to the observation of a pure 445 nm centred emission. Our interpretation, based on Siegel's work, is that the first shift is due to the adsorption of water and insertion in the carboxylate–zinc bond, bringing electron density toward the



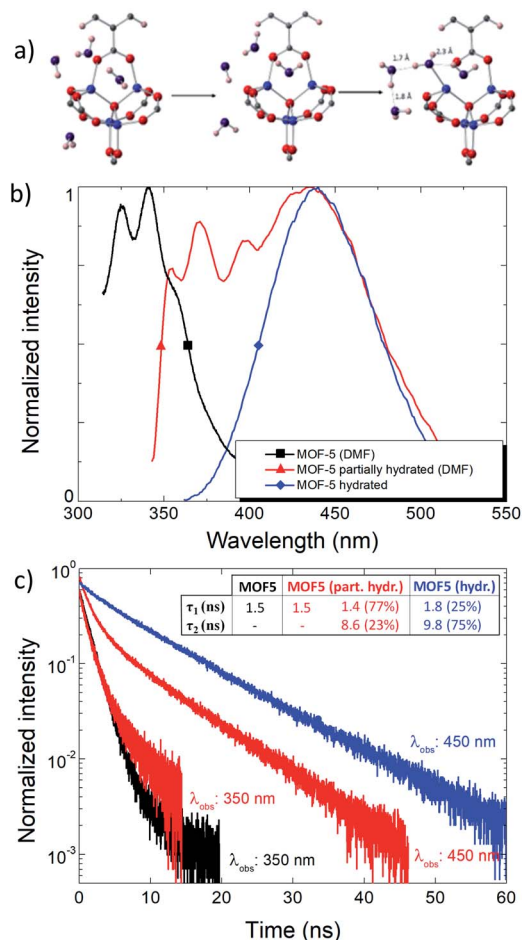


Fig. 4 (a) Adsorption and insertion of water in MOF. Print with permission from Siegel *et al.* from ref. 23 (b) PL emission data and (c) decay time of pure single crystal impregnated with DMF (black), partially hydrated single crystal impregnated with DMF (red) and fully hydrated (blue) fitting see ESI, Fig. S6.†

carboxylic group, resulting on shortening of bandgap. This could explain the displacement of fluorescence maximum towards longer wavelengths and emission broadening by molecular motion. The structure then collapses (totally or partially) due to the cleavage of Zn–O bonds (ESI, Fig. S3†). This collapse is also observed by gas adsorption analysis, with a decrease of the specific area from $2400 \text{ m}^2 \text{ g}^{-1}$ to $1800 \text{ m}^2 \text{ g}^{-1}$. At this point a complex of terephthalate– H_2O –zinc most probably creates the luminescence emitting at 445 nm. Our hypothesis is also based on TCSPC measurement (Fig. 4c). Decay times of partially hydrated MOF-5 (red) taken at 350 and 450 nm show a large discrepancy indicating at least two different emissive species. One has a very close resemblance to the fully hydrated MOF-5 (blue) emitting at 445 nm, whereas the faster component could be attributed to the crystalline MOF emitting at 345 nm. The loss of vibronic structure at 445 nm, due to hydration, attests from a luminescence mechanism that becomes different than a ligand centred emission and thus results in an elongation of lifetime and interpretation as a probable MLCT or LMCT transition.

In order to confirm this idea, Fig. 5 represents the degradation of MOF-5 by moisture exposure monitored by PL. The spectra show two distinct new emissions, the first one is assigned to the hydrated frameworks luminescence and the second one is assigned to a framework decomposition product, at 365 nm and 445 nm respectively. The general trend is that both emission intensities increase as well as water concentrates within the framework. Comparing normalized intensities highlights that the maximum of emission shifts from 355 nm to 365 nm before the framework decomposes, giving thus the awaited final 445 nm emission. The surface normalized intensity (ESI, Fig. S9†) shows an isobestic point at 410 nm confirming the transformation from one luminescent species to another one.

To conclude, millimetric MOF-5 single crystals have been successfully synthesized by solvothermal path. Structural characterizations were carried out showing highly pure phase materials in good agreement with literature precedents. Hence photophysical investigations using PL and decay time allow demonstrating that fluorescence in MOF-5 is ligand centred and therefore has a solvent guest dependent behavior. If free from solvent, the maximum emission occurs at 355 nm. This experimental value are in excellent agreement with excited states energy calculated by previous DFT studies. We also show that the integrity of the framework plays a major role in the resulting fluorescence. Indeed, hydration and structure collapse causes the broadening of the fluorescence and Zn₄O/BDC bond

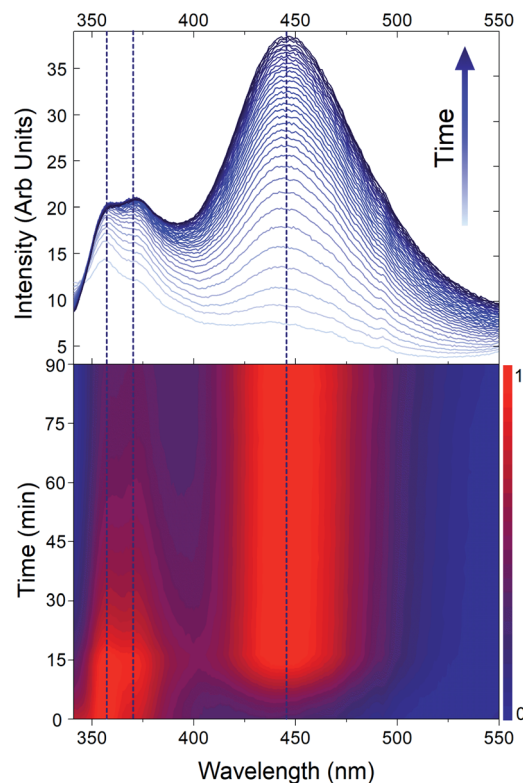


Fig. 5 Emission spectra of MOF-5 during time (top) and associate normalized intensity (bottom). Dashed lines represents different contributions.



cleavage by water promotes two new luminescent species with different emission wavelengths and longer lifetime than the original crystalline MOF. These observations tend to be similar for the isoreticular series with same topology as MOF-5 and will be explored. We expect that these results, which are an extension to the works already done on MOF-5, will have utility in comprehension on luminescent mechanisms involved in MOFs composed of Zn₄O building blocks.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 P. Silva, S. M. F. Vilela, J. P. C. Tomé and F. A. Almeida Paz, *Chem. Soc. Rev.*, 2015, **44**, 6774–6803.
- 2 M. D. Allendorf, C. A. Bauer, R. K. Bhakta and R. J. T. Houk, *Chem. Soc. Rev.*, 2009, **38**, 1330–1352.
- 3 Y. Cui, Y. Yue, G. Qian and B. Chen, *Chem. Rev.*, 2012, **112**, 1126–1162.
- 4 R. B. Lin, S. Y. Liu, J. W. Ye, X. Y. Li and J. P. Zhang, *Adv. Sci.*, 2016, **3**, 1–20.
- 5 W. P. Lustig, S. Mukherjee, N. D. Rudd, A. V. Desai, J. Li and S. K. Ghosh, *Chem. Soc. Rev.*, 2017, **46**, 3242–3285.
- 6 L. Wilbraham, F. X. Coudert and I. Ciofini, *Phys. Chem. Chem. Phys.*, 2016, **18**, 25176–25182.
- 7 T. D. Bennett, A. K. Cheetham, A. H. Fuchs and F. X. Coudert, *Nat. Chem.*, 2016, **9**, 11–16.
- 8 N. B. Shustova, T.-C. Ong, A. F. Cozzolino, V. K. Michaelis, R. G. Griffin and M. Dincă, *J. Am. Chem. Soc.*, 2012, **134**, 15061–15070.
- 9 S. Bordiga, C. Lamberti, G. Ricchiardi, L. Regli, F. Bonino, A. Damin, K. P. Lillerud, M. Bjørgen and A. Zecchina, *Chem. Commun.*, 2004, **10**, 2300–2301.
- 10 T. Tachikawa, J. R. Choi, M. Fujitsuka and T. Majima, *J. Phys. Chem. C*, 2008, **112**, 14090–14101.
- 11 P. L. Feng, J. J. Perry Iv, S. Nikodemski, B. W. Jacobs, S. T. Meek and M. D. Allendorf, *J. Am. Chem. Soc.*, 2010, **132**, 15487–15489.
- 12 M. Ji, X. Lan, Z. Han, C. Hao and J. Qiu, *Inorg. Chem.*, 2012, **51**, 12389–12394.
- 13 S. S. Kaye, A. Dailly, O. M. Yaghi and J. R. Long, *J. Am. Chem. Soc.*, 2007, **129**, 14176–14177.
- 14 O. Shekhah, H. Wang, D. Zacher, R. A. Fischer and C. Wöll, *Angew. Chem. Int. Ed.*, 2009, **48**, 5038–5041.
- 15 H. Li, M. Eddaoudi, M. O’Keeffe and O. M. Yaghi, *Nature*, 1999, **402**, 276–279.
- 16 J. Hafizovic, M. Bjørgen, U. Olsbye, P. D. C. Dietzel, S. Bordiga, C. Prestipino, C. Lamberti and K. P. Lillerud, *J. Am. Chem. Soc.*, 2007, **129**, 3612–3620.
- 17 H. Laguitton-Pasquier, R. Pansu, J. P. Chauvet, A. Collet, J. Faure and R. Lapouyade, *Chem. Phys.*, 1996, **212**, 437–455.
- 18 N. Mataga, Y. Kaifu and M. Koizumi, *Bull. Chem. Soc. Jpn.*, 1956, **29**, 465–470.
- 19 W. G. Shim, K. J. Hwang, J. T. Chung, Y. S. Baek, S. J. Yoo, S. C. Kim, H. Moon and J. W. Lee, *Adv. Powder Technol.*, 2012, **23**, 615–619.
- 20 J. E. Mondloch, O. Karagiari, O. K. Farha and J. T. Hupp, *CrystEngComm*, 2013, **15**, 9258–9264.
- 21 H. Q. Pham and T. Mai, *J. Phys. Chem. C*, 2014, **118**, 4567–4577.
- 22 J. Cepeda and A. Rodríguez-diéguez, *CrystEngComm*, 2016, **18**, 8556–8573.
- 23 Y. Ming, N. Kumar and D. J. Siegel, *Omega*, 2017, **2**, 4921–4928.
- 24 J. A. Greathouse and M. D. Allendorf, *J. Am. Chem. Soc.*, 2006, **128**, 10678–10679.

