


Cite this: *RSC Adv.*, 2020, 10, 17713

# Dip-coating decoration of Ag<sub>2</sub>O nanoparticles on SnO<sub>2</sub> nanowires for high-performance H<sub>2</sub>S gas sensors

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SnO<sub>2</sub> nanowires (NWs) are used in gas sensors, but their response to highly toxic gas H<sub>2</sub>S is low. Thus, their performance toward the effective detection of low-level H<sub>2</sub>S in air should be improved for environmental-pollution control and monitoring. Herein, Ag<sub>2</sub>O nanoparticle decorated SnO<sub>2</sub> NWs were prepared by a simple on-chip growth and subsequent dip-coating method. The amount of decorated Ag<sub>2</sub>O nanoparticles on the surface of SnO<sub>2</sub> NWs was modified by changing the concentration of AgNO<sub>3</sub> solution and/or dipping times. Gas-sensing measurements were conducted at various working temperatures (200–400 °C) toward different H<sub>2</sub>S concentrations ranging within 0.1–1 ppm. The selectivity of Ag<sub>2</sub>O-decorated SnO<sub>2</sub> NW sensors for ammonia and hydrogen gases was tested. Results confirmed that the Ag<sub>2</sub>O-decorated SnO<sub>2</sub> NW sensors had excellent response, selectivity, and reproducibility. The gas-sensing mechanism was interpreted under the light of energy-band bending by sulfuration, which converted the p–n junction into n–n, thereby significantly enhancing the sensing performance.

Received 11th March 2020

Accepted 30th April 2020

DOI: 10.1039/d0ra02266g

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## 1. Introduction

Air pollution caused by H<sub>2</sub>S gas is extremely dangerous even at low concentrations (sub-ppm level) because this gas is colorless, flammable, and highly toxic.<sup>1</sup> The sources of H<sub>2</sub>S are very diverse<sup>2</sup> because it can be produced naturally from crude petroleum, oil drilling, and volcano eruption or from the bacterial decomposition of organic matter in anaerobic environments.<sup>3</sup> H<sub>2</sub>S is also produced as a by-product in biogas plants during waste treatment.<sup>4</sup> The effects of H<sub>2</sub>S on the human body are summarized in Table 1.<sup>5</sup> The threshold odor concentration of H<sub>2</sub>S is about 10 ppb, but its toxic concentration range is very broad (*i.e.*, from ppb to ppm). The threshold limit of H<sub>2</sub>S is reportedly 0.003 ppm for 8 h of exposure.<sup>5</sup> However, the permissible concentration of H<sub>2</sub>S recommended by the Scientific Advisory Board on Toxic Air Pollutants (USA) ranges within 20–100 ppb.<sup>6</sup> Thus, effective gas sensors for detecting low levels of H<sub>2</sub>S under field conditions are urgent to develop.<sup>2</sup>

Many techniques for H<sub>2</sub>S detection have been developed, but metal oxide-based resistive-type gas sensors are advantageous because of their low cost, high sensitivity, real-time detection, portability, and low power consumption.<sup>7–10</sup> SnO<sub>2</sub> (ref. 11) is one

of the most popular materials for such sensors because of its relatively high sensitivity to various gases, as well as its feasibility in functionalization to improve sensing performance.<sup>12,13</sup> However, SnO<sub>2</sub> has the main drawback of low response to low concentration of H<sub>2</sub>S<sup>14</sup> and poor selectivity over air-polluting gases such as NH<sub>3</sub>, H<sub>2</sub>S, and CO.<sup>15,16</sup> This problem can be solved by using heterojunctions between two dissimilar semiconducting materials, which utilizes the unique effects and leads to enhanced sensor performance.<sup>15</sup> Nano-heterostructures are often utilized owing to their small size and high surface-to-volume ratio,<sup>17,18</sup> and many efforts have been devoted to the fabrication of p–n heterojunctions for increasing H<sub>2</sub>S-sensing performance.<sup>19–24</sup> The most common p-type metal oxides used to form heterojunctions with n-type SnO<sub>2</sub> semiconductor are CuO,<sup>18,25</sup> NiO,<sup>26,27</sup> and Co<sub>3</sub>O<sub>4</sub> (ref. 24) because of their easy sulfidation into CuS, NiS, and CoS, respectively. However, sensors with these oxides can detect H<sub>2</sub>S gas only at high concentrations of >10 ppm<sup>28</sup> because the sulfidation of transition-metal oxides requires a high supply of sulfur source.<sup>29</sup> Meanwhile, Ag<sub>2</sub>O has unique characteristics that enable it to functionalize SnO<sub>2</sub> nanomaterials to enhance gas-sensing performance to different gases such as H<sub>2</sub>,<sup>30</sup> ethanol,<sup>31</sup> and CO.<sup>32</sup> The decoration of p-type Ag<sub>2</sub>O on the surface of n-type SnO<sub>2</sub> is advantage over the use of metallic Ag because it forms the p–n heterojunction, thus enhances the gas sensing performance.<sup>33–35</sup> Ag<sub>2</sub>O is also reported easily converted into Ag<sub>2</sub>S in the presence of H<sub>2</sub>S<sup>36</sup> because of its low free Gibbs energy for the reaction. The free Gibbs energy for conversion of Ag<sub>2</sub>O, CuO, and NiO into Ag<sub>2</sub>S,

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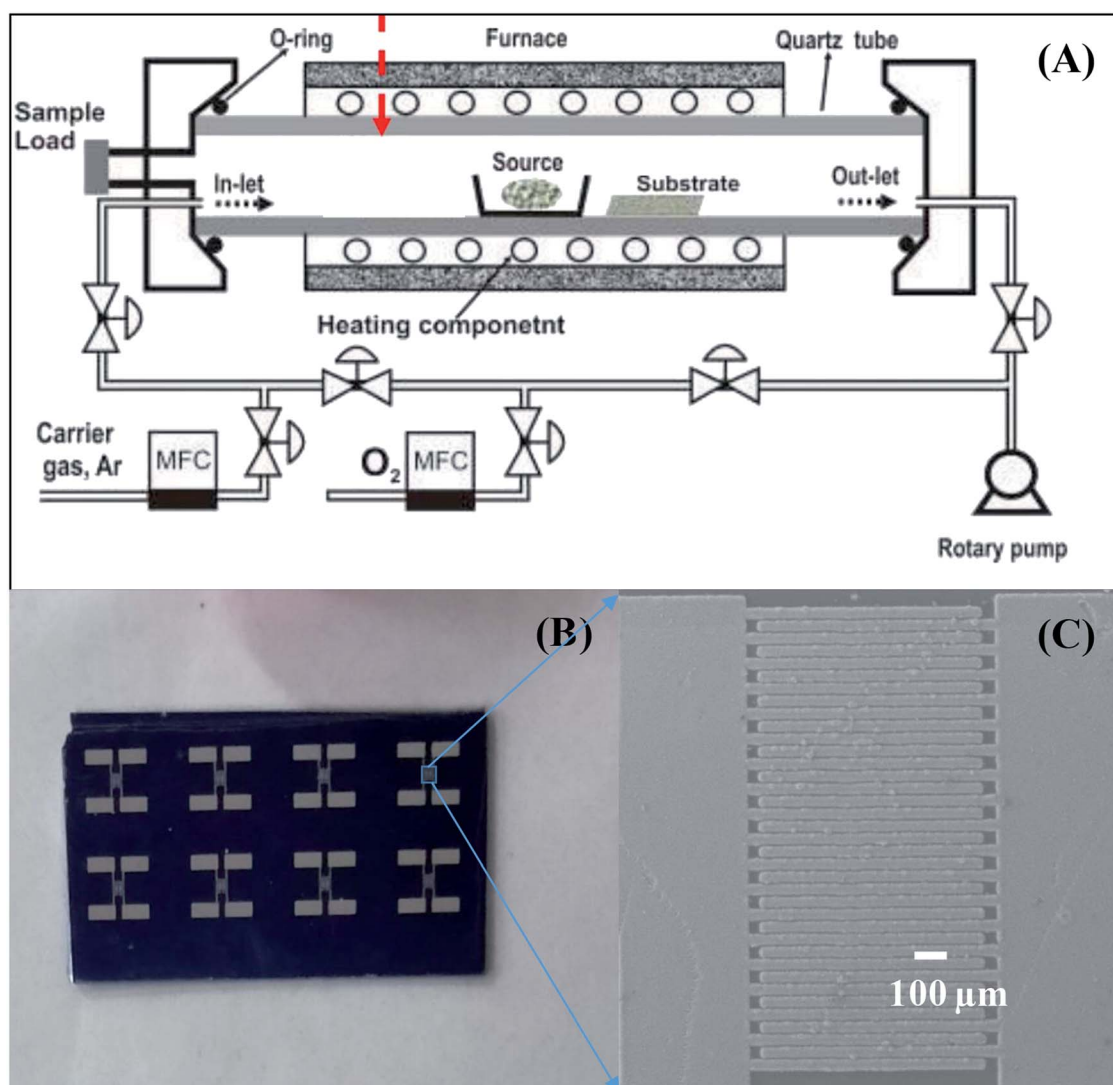


Table 1 Effects of exposure to H<sub>2</sub>S.<sup>5</sup>

Concentration (ppm)	Effects
0.003–0.02	Approximate threshold for odor
3–10	Obvious offensive odor
50–100	Serious eye irritation and respiratory tract irritation
100–200	Loss of smell
250–500	Fluid buildup in lungs and imminent threat to life
500	Anxiety, headache, dizziness, excessively rapid respiration, amnesia, and unconsciousness
500–1000	Immediate collapse, irregular heartbeat, neural paralysis, and respiratory paralysis leading to death

CuS, and NiS in the present of H<sub>2</sub>S gas is  $-224.7$ ,  $-119.1$ , and  $-62.5$  kJ mol<sup>-1</sup>, respectively. Therefore, decoration of Ag<sub>2</sub>O nanoparticles on the surface of SnO<sub>2</sub> is expected to show better sensing performance such as low detection limit of H<sub>2</sub>S with higher sensitivity than others. However, few studies have focused on improving of H<sub>2</sub>S-sensing properties using Ag<sub>2</sub>O/SnO<sub>2</sub> thin film.<sup>33–35</sup> It is hard to find the related work reported on

the decoration of Ag<sub>2</sub>O on the surface of SnO<sub>2</sub> NWs for enhanced H<sub>2</sub>S gas despite the significantly higher stability of NWs than their thin-film counterparts.<sup>37</sup> Doped thick films have shown good sensitivity to low concentrations of H<sub>2</sub>S but are not feasible to miniaturize.<sup>33</sup> Decorated thin films present poor response to high concentrations of H<sub>2</sub>S.<sup>34</sup> A previous work<sup>35</sup> has reported extremely low response (99%) to the high H<sub>2</sub>S



Scheme 1 Sensor fabrication process: (A) CVD system used to grow SnO<sub>2</sub> NWs, (B) photo of sensor chips; (C) SnO<sub>2</sub> NW sensor after fabrication.



concentration of 50 ppm at 74 °C. Our group has recently reported the H<sub>2</sub>S-sensing characteristics of self-heated Ag-coated SnO<sub>2</sub> NWs, where the decoration of Ag is realized by sputtering method.<sup>38</sup> However, this method requires vacuum conditions and expensive equipment for Ag decoration, and the content of Ag<sub>2</sub>O nanoparticles on the surface of SnO<sub>2</sub> NWs are difficult to control. Thus, a low-cost, suitable, and effective method for functionalizing p-type Ag<sub>2</sub>O nanoparticles with low activation energy for reversible sulfidation and oxidation, as well as enhanced H<sub>2</sub>S-sensing performance of SnO<sub>2</sub> NWs, must be developed.

Herein, we reported the dip-coating decoration of Ag<sub>2</sub>O nanoparticles on the surface of on-chip-grown SnO<sub>2</sub> NWs to enhance their H<sub>2</sub>S gas-sensing performance. Decoration was realized by dipping the sensor in AgNO<sub>3</sub> solution, followed by oxidation to form Ag<sub>2</sub>O nanocrystals on the surface of SnO<sub>2</sub> NWs. The effects of Ag<sub>2</sub>O content on the H<sub>2</sub>S gas-sensing performance of the SnO<sub>2</sub> NWs were studied to maximize sensor response to H<sub>2</sub>S. Results demonstrated that the sensors processed excellent performance for monitoring extremely low H<sub>2</sub>S concentrations. The H<sub>2</sub>S gas-sensing mechanism of the SnO<sub>2</sub> NWs functionalized with Ag<sub>2</sub>O nanoparticles was also discussed through the perspective of band-structure and sulfurization process.

## 2. Experimental

The preparation of SnO<sub>2</sub> NWs-based sensors has been described in our previous publication.<sup>14</sup> The NW sensors were directly grown on thermally oxidized silicon substrate using a chemical vapor deposition system, as shown in Scheme 1(A).<sup>39</sup> In a typical procedure, SnO<sub>2</sub> NWs were grown on seeded Pt electrodes at 750 °C from a starting material of Sn powder through thermal evaporation. Growth proceeded at 750 °C for 20 min with an oxygen gas flow of 0.5 sccm and pressure of  $1.8 \times 10^{-1}$  torr. For one batch of fabrication, up to 8 sensors were obtained, as shown in Scheme 1(B). The SnO<sub>2</sub> NWs were homogeneously grown on the Pt electrode fingers, as shown in Scheme 1(C). The bare SnO<sub>2</sub> NWs sensors were decorated with Ag<sub>2</sub>O nanoparticles by dip coating in AgNO<sub>3</sub> solutions and subsequent annealing at 500 °C for 3 h in air. This decoration method had the advantage over the sputtering method of not requiring vacuum conditions.<sup>38</sup> The density of Ag<sub>2</sub>O nanoparticles decorated on the surface of SnO<sub>2</sub> NWs was controlled by varying the concentration of AgNO<sub>3</sub> solution (0.05, 0.2, and 1 mM) and the dipping times (1, 5, and 20 times). The samples were denoted as S0, S1, S2, S3, S4, and S5 (Table 2). The morphology, chemical composition and structural characteristics of pristine and Ag<sub>2</sub>O-decorated SnO<sub>2</sub> NWs were investigated by scanning electron microscopy (SEM; JEOL 7600F), energy-dispersive X-ray spectroscopy (EDS), high-resolution transmission electron microscopy (HRTEM; JEOL 2100F), and X-ray diffraction (XRD; D8 Advance).<sup>3</sup>

Gas-sensing properties were measured using a SourceMeter® Keithley 2602B. Details about the gas-sensing measurement system are described elsewhere.<sup>40</sup> Dry air was used as reference and diluting gas. Sensor response to different

Table 2 Samples at different AgNO<sub>3</sub> concentrations and dipping times

Sample	AgNO <sub>3</sub> concentration (mM)	Dipping times
S0	0	0
S1	0.05	1
S2	0.2	1
S3	1	1
S4	1	5
S5	1	20

H<sub>2</sub>S concentrations (0.1–1 ppm) at various working temperatures (200, 250, 300, 350, and 400 °C) were investigated. The selectivity among reducing gases (including ammonia and hydrogen) and the reproducibility of the sensors were also tested. During gas-sensing measurements, sensor resistance was continuously recorded, and the target gas and dry air were alternatively switched on/off. Gas response was defined as  $S = R_a/R_g$  for the reducing gas H<sub>2</sub>S, where  $R_a$  and  $R_g$  are the sensor resistances in air and in target gas, respectively.<sup>3</sup>

## 3. Results and discussion

### 3.1. Material characterization

We did not characterize all samples and instead selected sensors S1, S2, and S5 for SEM, EDS, and TEM analysis. Fig. 1(A) illustrates a SEM image of SnO<sub>2</sub> NWs (S1) grown on patterned Pt electrodes. Notably, the electrode finger was 20 μm wide [inset of Fig. 1(A)]. Although the gap between two electrode fingers was 20 μm, the grown SnO<sub>2</sub> NWs can still efficiently cover the gaps, as shown in the inset of Fig. 1(A). SnO<sub>2</sub> NWs grew primarily on the surface of Pt electrode fingers, but their lengths were controlled sufficiently to connect between the fingers and thus act as conducting channels in the gas-sensing measurement. The average diameter of SnO<sub>2</sub> NWs was approximately 70 nm. The surface of pristine SnO<sub>2</sub> NWs was as smooth as that of the single crystal. This result was consistent with the growth of SnO<sub>2</sub> NWs by vapor–liquid–solid mechanism.<sup>41</sup> Herein, we did not use Au as catalyst during the growth of SnO<sub>2</sub> NWs, so belt-like NWs were obtained at the initial state. A SnO<sub>2</sub> NW comprises a single crystal, as reported in our previous article.<sup>14</sup> Composition analysis of the SnO<sub>2</sub> NW by EDS [Fig. 1(B)] revealed the existence of O, Sn, and Pt elements. Pt was originally from the electrode, whereas O and Sn were from the SnO<sub>2</sub> NWs.

The SEM image of SnO<sub>2</sub> NWs after decoration with Ag<sub>2</sub>O nanoparticles (S2) is presented in Fig. 1(C), whose inset is a low-magnification SEM image. The electrode fingers were covered by the SnO<sub>2</sub> NWs. Ag<sub>2</sub>O decoration by dip coating maintained the morphology of the SnO<sub>2</sub> NWs, but their surface was not as smooth as that of the pristine sample and tiny particles can be seen in the SEM images. The high-magnification SEM image revealed the presence of Ag<sub>2</sub>O nanoparticles on the surface of SnO<sub>2</sub> NWs. EDS composition analysis of S2 [Fig. 1(D)] confirmed the presence of Ag at an energy of 2.98 eV despite the quantitative evaluation displaying a value of zero.



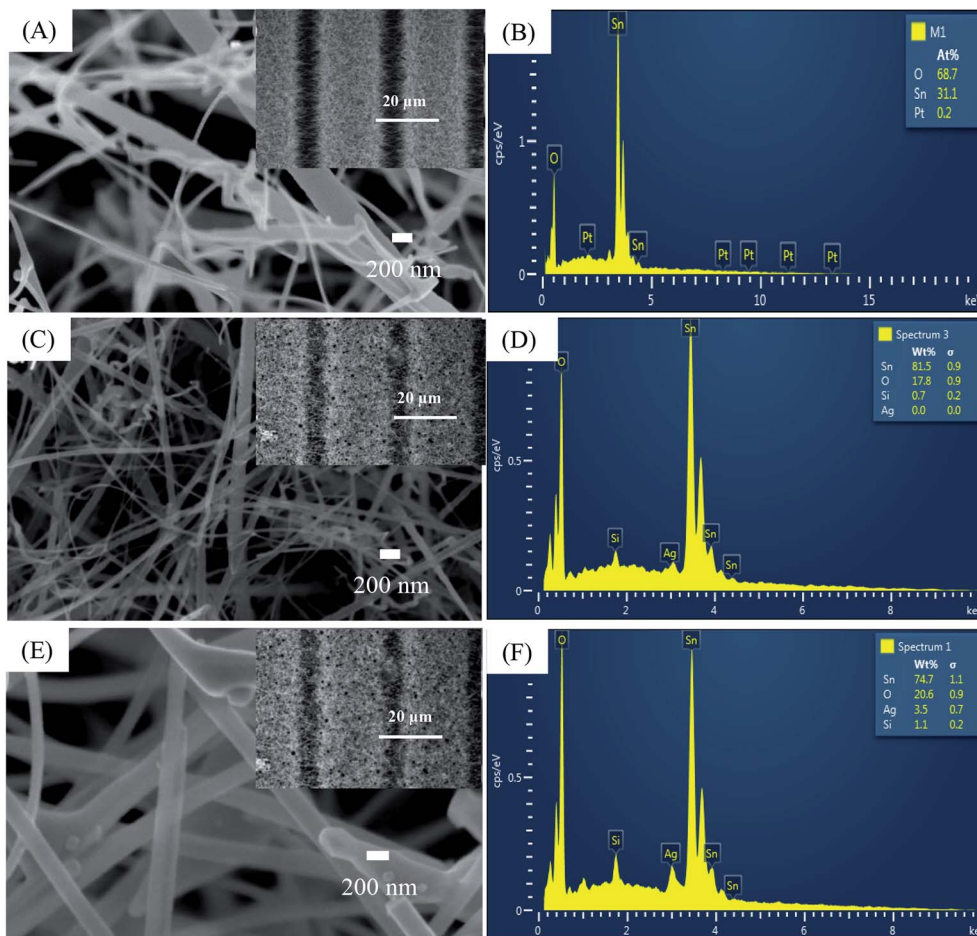


Fig. 1 SEM images and EDS analysis of: (A and B) sensor S1 (0.05 mM); (C and D) sensor S2 (0.2 mM), (E and F) sensor S5 (1 mM). Insets are correspondent sensors' fringes.

The SEM image of S5 is shown in Fig. 1(E), whose inset is a low-magnification SEM image of S5. With increased  $\text{AgNO}_3$  amount in dipping solution and dipping times, the morphology of the  $\text{SnO}_2$  NWs slightly changed. More tiny particles can be seen in the SEM image of S5, but the sample maintained its entangled NW morphology. Whether the  $\text{Ag}_2\text{O}$  nanoparticles continuously or discontinuously decorated the surface of  $\text{SnO}_2$  NWs was difficult to observe simply by SEM observation. However, the surface of the samples was found to have increased roughness with increased  $\text{Ag}_2\text{O}$  decoration. EDS composition analysis of S5 [Fig. 1(F)] showed that the content of Ag was very high (about 3.5 wt%). This result demonstrated that increasing the concentration of  $\text{AgNO}_3$  solution and the dipping times can increase the content of  $\text{Ag}_2\text{O}$  nanoparticles decorated on the surface of  $\text{SnO}_2$  NWs for effective  $\text{H}_2\text{S}$  detection.

To further study the decoration of  $\text{Ag}_2\text{O}$  on the surface of  $\text{SnO}_2$  NWs, we selected S1, S2, and S5 for TEM characterizations. The grown  $\text{SnO}_2$  NWs had a very smooth and clean surface [Fig. 2(A)]. The average diameter of a  $\text{SnO}_2$  NW was approximately 70 nm, consistently with the observation by SEM images. No  $\text{Ag}_2\text{O}$  nanoparticle was observed in this sample possibly because the  $\text{AgNO}_3$  concentration of the dipping solution was too low. The HRTEM images of S2 and S5 are shown in Fig. 2(B), and (C), respectively. The black dots decorated on the  $\text{SnO}_2$  NWs

surface were  $\text{Ag}_2\text{O}$  nanoparticles. Given that S2 was decorated by a low concentration of  $\text{AgNO}_3$  (0.2 mM) solution, the density of  $\text{Ag}_2\text{O}$  nanoparticles on the surface of  $\text{SnO}_2$  NWs was very low [Fig. 2(B)] and the particle sizes were about 7 nm. The size and density of  $\text{Ag}_2\text{O}$  increased with increased  $\text{AgNO}_3$  concentration (1 mM, 20 times of dipping), as observed in S5 [Fig. 2(C)]. The diameter of  $\text{Ag}_2\text{O}$  nanoparticles decorated on the surface of  $\text{SnO}_2$  NWs ranged within 5–20 nm. However, they were still smaller than the diameter of  $\text{SnO}_2$  NWs. The wettability of  $\text{AgNO}_3$  solution on the  $\text{SnO}_2$  NW surface is very important for the decoration of  $\text{Ag}_2\text{O}$  nanoparticles performed by dipping method. Wettability ensures the homogenous decoration of  $\text{Ag}_2\text{O}$  nanoparticles on the total surface of  $\text{SnO}_2$  NWs. At a high density,  $\text{Ag}_2\text{O}$  particles may agglomerate and form a large cluster. A high-magnification HRTEM image of about 5 nm  $\text{Ag}_2\text{O}$  nanoparticle is shown in Fig. 2(D). The interspacing of  $\sim 0.23$  nm, which corresponded to the (200) lattice plane of cubic structured  $\text{Ag}_2\text{O}$ ,<sup>42</sup> was observed. This result was consistent with a previous one on the thermal decomposition of  $\text{AgNO}_3$  at 250–440 °C (ref. 43) into Ag. Then, Ag was oxidized into  $\text{Ag}_2\text{O}$  at an oxidation temperature of about 350–500 °C.<sup>44</sup> In the process of e-beam decoration, Ag nanoparticles are anisotropically decorated on one side of NWs but not homogeneously.<sup>41</sup> Herein, the wet chemical method was used to ensure





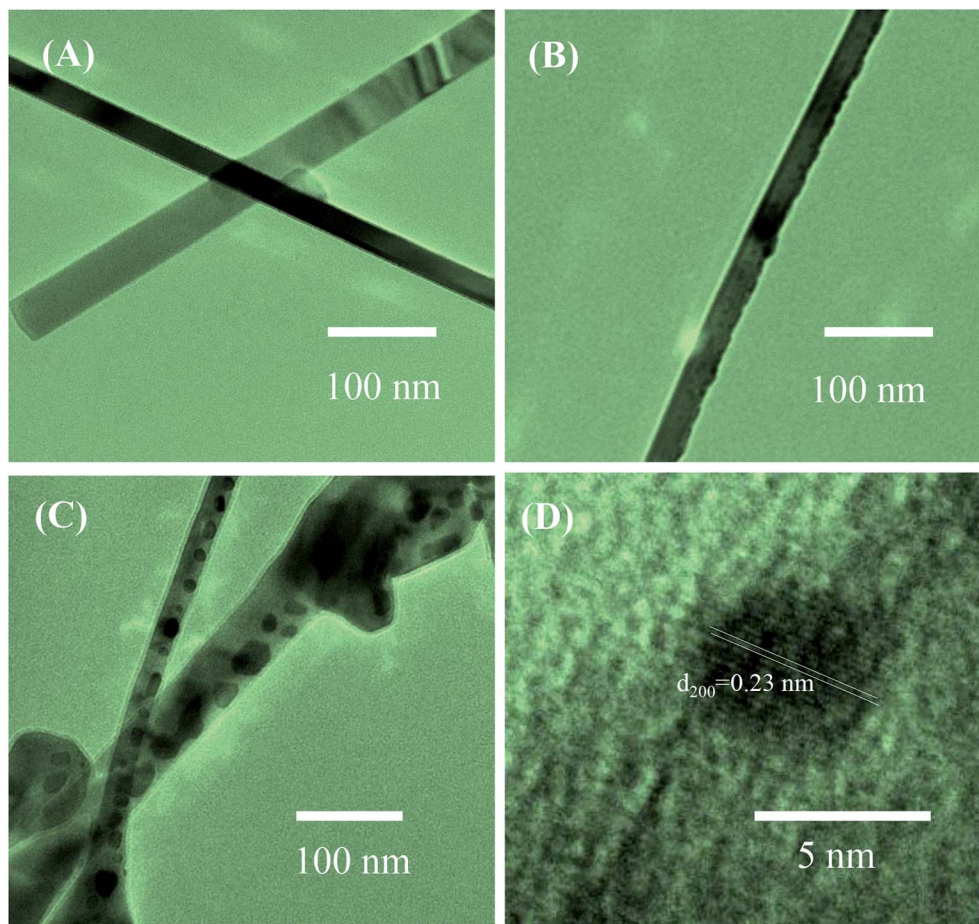


Fig. 2 TEM images of  $\text{SnO}_2$  nanowires and  $\text{SnO}_2$  nanowires decorated with  $\text{Ag}_2\text{O}$  nanoparticles: (A) sensor S1; (B) sensor S2, (C) sensor S5; (D) HRTEM image of  $\text{Ag}_2\text{O}$  nanoparticle on the surface of nanowire.

that nanoparticles were homogeneously decorated on the surface of the NWs. Notably, S5 had larger  $\text{Ag}_2\text{O}$  nanoparticles than S2, but decoration was not continuous because over-decoration of  $\text{Ag}_2\text{O}$  nanoparticles can reduce sensor response.<sup>45</sup>

### 3.2. Gas-sensing characteristics

Fig. 3(A)–(F) show the changes in transient resistance with time of S0–S5, respectively, upon exposure to various  $\text{H}_2\text{S}$  concentrations (0.1, 0.25, 0.5, and 1 ppm) measured at different working temperatures (200, 250, 300, 350, and 400 °C). S0 showed significant response to  $\text{H}_2\text{S}$  at all measured temperatures, but the response and recovery times were very long at low working temperature [Fig. 3(A)]. At a working temperature of 200 °C, S0 required almost 1.5 h to finish one measurement at four concentrations of  $\text{H}_2\text{S}$ . Thus, stair-type tests were conducted for  $\text{H}_2\text{S}$  gas sensing because of the slow recovery characteristics [Fig. 3(B)–(F)]. This finding indicated that measurements were conducted through a stepwise increase in  $\text{H}_2\text{S}$  concentration from 0.1 ppm to 1 ppm before finally being refreshed by dry air. The obtained plots illustrated that the resistance of pristine and decorated  $\text{SnO}_2$  NW sensors steeply increased when  $\text{H}_2\text{S}$  gas was injected into the test chamber [Fig. 3(B)–(F)]. The resistance then recovered to the initial values

when  $\text{H}_2\text{S}$  was replaced by dry air. All these sensors presented the typical n-type gas-sensing behavior of  $\text{SnO}_2$  NW semiconductor, where resistance decreased with increased  $\text{H}_2\text{S}$  gas exposure. The base resistance in air of pristine  $\text{SnO}_2$  NWs (S0) was much smaller than that of  $\text{Ag}_2\text{O}$ -decorated  $\text{SnO}_2$  sensors from S1 to S5. S5, with the largest amount of  $\text{Ag}_2\text{O}$  decoration, had the highest resistance values in air of about 7 M $\Omega$  at 200 °C. Notably,  $\text{Ag}_2\text{O}$  is also a good conductor, so the high base resistance value of S5 confirmed that the nanoparticles decorated on the surface of  $\text{SnO}_2$  NW formed the p–n heterojunction. Based on the plot of transient resistance *versus* time of the sensors, we roughly estimated that the response values increased but the recovery rate of the sensors decreased with increased  $\text{Ag}_2\text{O}$  decoration.

The quantitative response values of different sensors are shown in Fig. 4(A)–(F). The response values of all sensors decreased with increased working temperature within the measured range. This result was similar to that of other metal-oxide-based  $\text{H}_2\text{S}$  gas sensors.<sup>46</sup> The pristine  $\text{SnO}_2$  NW sensor (S0) had the highest response value of less than 4 over all the range of working temperatures and gas concentrations [Fig. 4(A)]. The response values for 1 ppm  $\text{H}_2\text{S}$  decreased almost linearly from 3.6 to 2.9 with increased working temperature

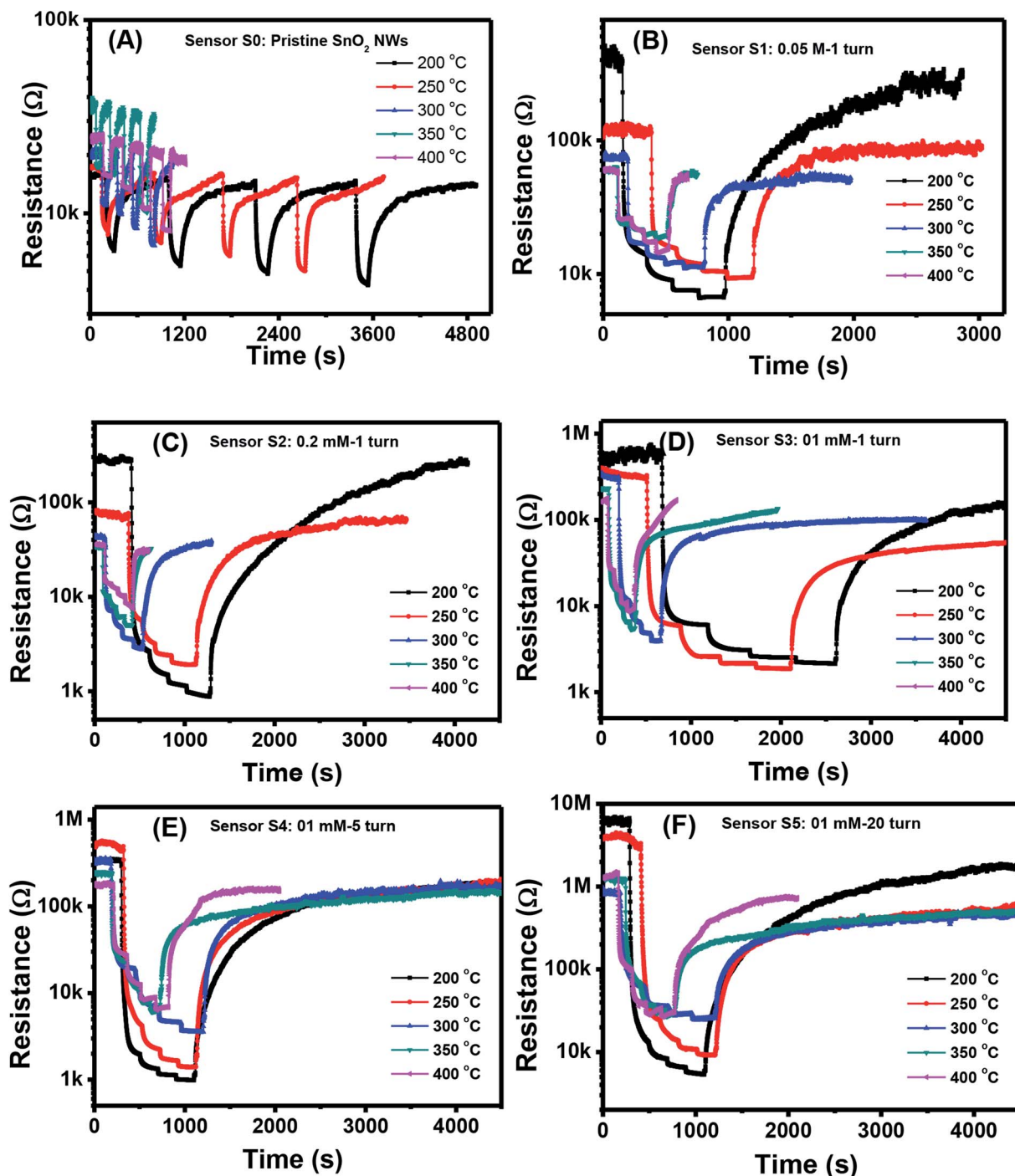


Fig. 3 Transient response to different  $\text{H}_2\text{S}$  concentrations various working temperature of the sample: S0 (A), S1 (B), S2 (C), S3 (D), S4 (E), S5 (F).

from 200 °C to 400 °C. These values were very low compared with the response of  $\text{Ag}_2\text{O}$ -decorated  $\text{SnO}_2$  NW sensors [Fig. 4(B)–(F)]. The responses of  $\text{Ag}_2\text{O}$ -decorated  $\text{SnO}_2$  NW sensors from S1 to S5 were much higher than that of pristine  $\text{SnO}_2$  NWs, S0. The responses of  $\text{Ag}_2\text{O}$ -decorated  $\text{SnO}_2$  NW sensors increased with increased amount of decorated  $\text{Ag}_2\text{O}$  nanoparticles. All sensors showed better response at lower operating temperatures, reaching the highest values at 200 °C within the measured range. The response values to 1 ppm  $\text{H}_2\text{S}$

at 200 °C of S1–S5 were approximately 61, 358, 392, 690 and 1155, respectively. The response to 1 ppm  $\text{H}_2\text{S}$  at 200 °C of S5 was about 320-fold higher than that of S0 under the same measurement condition. Notably, the maximum response to 1 ppm  $\text{H}_2\text{S}$  gas of rGO-loaded  $\text{Fe}_2\text{O}_3$  nanofibers is only 9.2.<sup>46</sup> Herein, all sensors had decreased response values with increased working temperature from 200 °C to 400 °C. The response to 0.1 ppm  $\text{H}_2\text{S}$  at 400 °C of S1 was 2.5, whereas that of S5 was much higher at about 16. Clearly, within the studied  $\text{H}_2\text{S}$



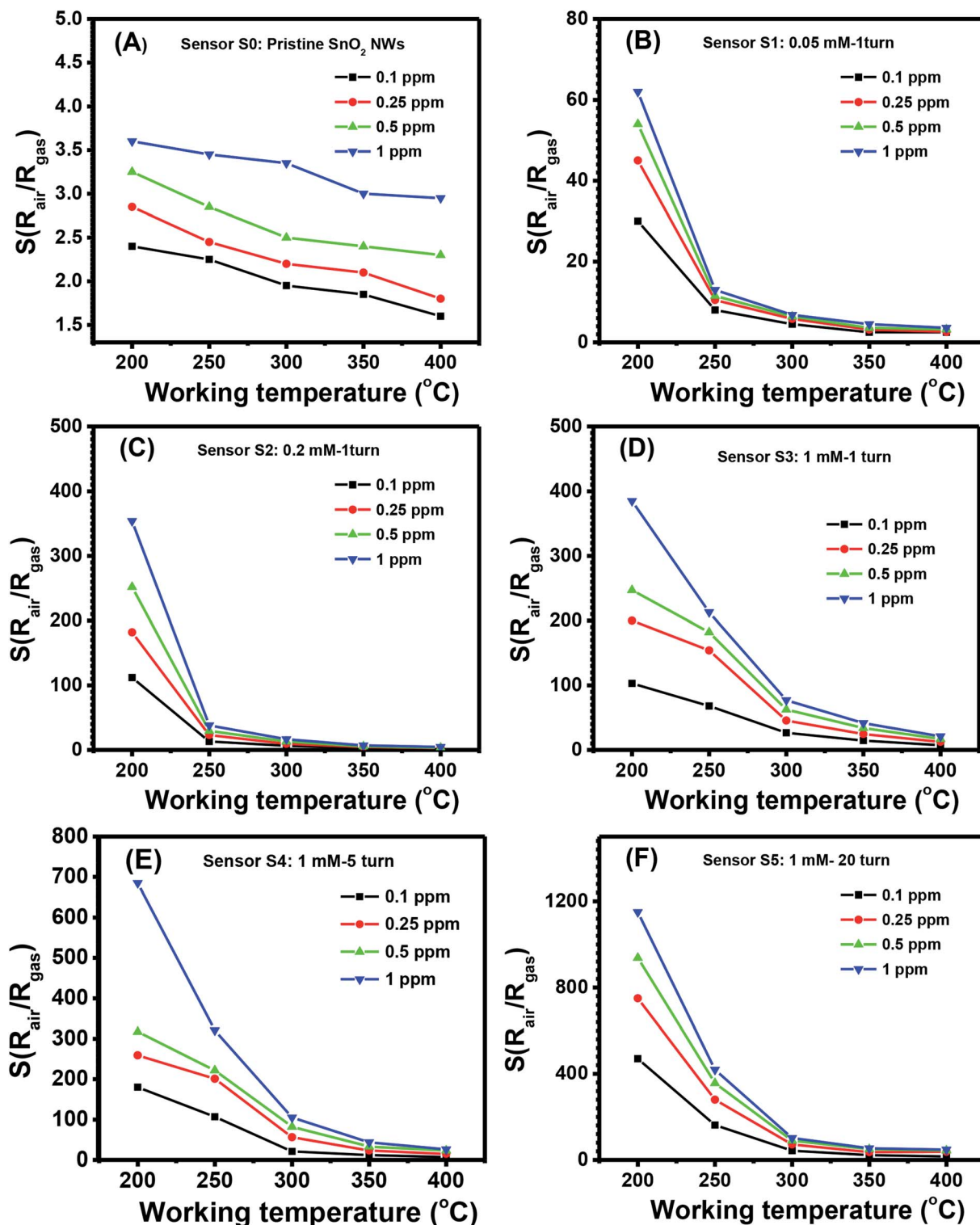


Fig. 4 Response to different  $H_2S$  concentrations various working temperature of the sample: S0 (A), S1 (B), S2 (C), S3 (D), S4 (E), S5 (F).

concentration range (0.1–1 ppm), S5 had the best sensing performance because of its high sensitivity. This finding can be attributed to the p–n heterojunctions between  $Ag_2O$  nanoparticles on the surface and  $SnO_2$  NWs,<sup>31,47</sup> similar to the p–n heterojunctions between  $CuO$  and  $SnO_2$  (ref. 19) or  $NiO$  and

$SnO_2$ .<sup>21</sup> Details about the gas-sensing mechanism are discussed in subsequent sections.

The response values of different sensors measured at 200 °C as a function of  $H_2S$  concentration are shown in Fig. 5(A). With a low content of  $Ag_2O$  decoration (S0 to S4), the response values increased almost linearly with  $H_2S$  concentration in the



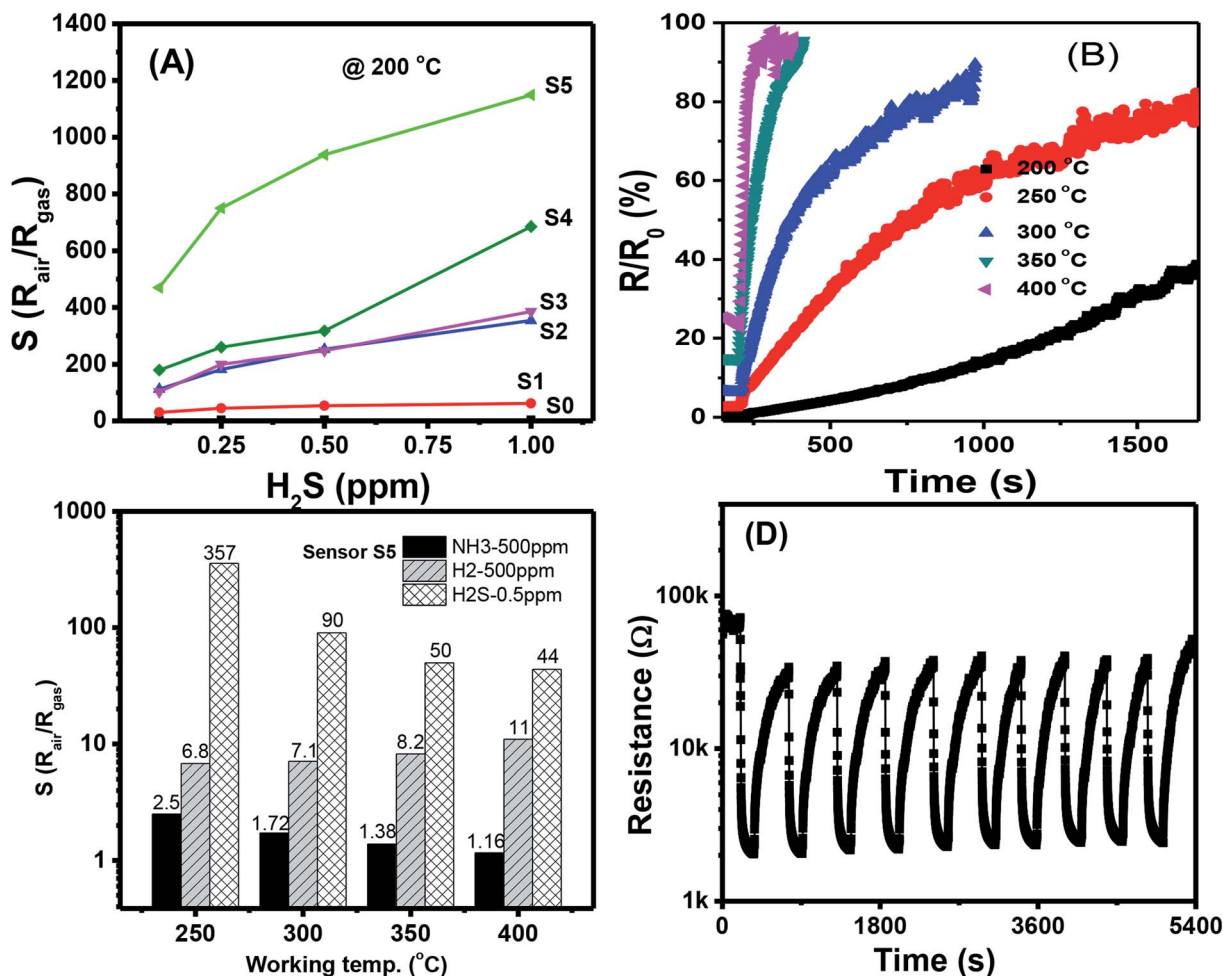


Fig. 5 (A) Comparative response of different sensors at 200 °C; (B) response time of sensor S5 measured at different temperatures; (C) response of the sensor S5 at different working temperatures to various gases; (D) stability of the sensor S5 for 10 cycles measured at 250 °C.

measured range, but their values were low. The highest response value of the sensor S0 at 200 °C for 1 ppm  $\text{H}_2\text{S}$  is about 3.7. S5 had the highest response values, and the response values increased nonlinearly with  $\text{H}_2\text{S}$  concentration. Along with the gas response, the recovery time of the sensor is very important in practical applications because it determines sensor reusability. The effects of working temperature on the recovery time of  $\text{Ag}_2\text{O}$ -decorated  $\text{SnO}_2$  NW sensor are shown in Fig. 5(B). Obviously, the sensor had very poor recovery characteristics at low working temperatures of 200, 250, and 300 °C, *i.e.*, resistance did not recover to the initial value after refreshing for 1000 s. However, the sensor presented 100% recovery characteristics at working temperatures of 350 and 400 °C, with a recovery time of approximately 70 s. In practical application, balance should be achieved between sensor sensitivity and recovery depending on the objective of the application. For instance, the sensors based on 2D materials have poor recovery characteristics, but they could operate at room temperature, thus suitable for low power consumption devices.<sup>48,49</sup> Herein, the long recovery time is possible due to the formation enthalpy of  $\text{Ag}_2\text{S}$  ( $-32.6 \text{ kJ mol}^{-1}$ ) is lower than that of  $\text{Ag}_2\text{O}$

( $-31 \text{ kJ mol}^{-1}$ ), thus it requires higher energy to break the bonding of  $\text{Ag}_2\text{S}$  than that of  $\text{Ag}_2\text{O}$  compound. As a result, the sensor has longer recovery time than the response time. The selectivity of S5 toward three reducing gases  $\text{H}_2\text{S}$ ,  $\text{NH}_3$ , and  $\text{H}_2$  was tested, and the results are shown in Fig. 5(C). At a low working temperature of 200 °C, the sensor did not show good recovery to  $\text{H}_2\text{S}$ , so we tested the selectivity at 250, 300, 350, and 400 °C. Results demonstrated that S5 had the highest response toward 0.5 ppm  $\text{H}_2\text{S}$  despite the 1000-fold concentration in all working temperatures. At a working temperature of 400 °C, S5 still had a high response value of 44–0.5 ppm  $\text{H}_2\text{S}$ , whereas the corresponding values for 500 ppm  $\text{NH}_3$  and 500 ppm  $\text{H}_2$  were 1.16 and 11, respectively. Reproducibility and repeatability are also important properties of a gas sensor; thus, we tested the short-term stability of the sensor by switching on/off the ambient from air to 0.25 ppm  $\text{H}_2\text{S}$  gas and back to air at a working temperature of 250 °C. As shown in Fig. 5(D), excepted for the first cycle, the sensor exhibited good recovery characteristics for 10 pulses of measurement, where the base resistance recovered to the initial value after refreshing the chamber with air. The relative standard deviation (RSD) was





**Table 3** A comparative result on the functionalized-SnO<sub>2</sub> gas sensors for H<sub>2</sub>S detection

Material	Conc. (ppm)	Working temp (°C)	Response ( $R_a/R_g$ )	Response/recovery times (s)	Ref.
Ag <sub>2</sub> O-SnO <sub>2</sub> thin film	50	74	99 <sup>a</sup>	>600/4500	35
CuO-SnO <sub>2</sub> NWs	80	300	1280	1/828	23
NiO-SnO <sub>2</sub> NWs	10	300	1372	11/102	21
CuO-SnO <sub>2</sub> NWs	10	250	26.3	180/600	19
CuO-SnO <sub>2</sub> nanofibers	10	150	3000	2/3000	25
CuO-SnO <sub>2</sub> thin film	100	180	25.3	10/42	22
CuO-SnO <sub>2</sub> hollow spheres	1	300	22.4	500/1000	50
Ag <sub>2</sub> O-SnO <sub>2</sub> mesoporous	1	100	71.5	390/1600	33
NiO-SnO <sub>2</sub> nanoweb	100	300	~6	N/A	26
NiO-SnO <sub>2</sub> thin film	10	Room	440	2000/30 000	27
Ag <sub>2</sub> O-SnO <sub>2</sub> NWs	0.5	—	21	12/1000	51
SnO <sub>2</sub> NWs	1	—	~3.5	50/200	14
<b>Ag<sub>2</sub>O-SnO<sub>2</sub> NWs</b>	<b>1</b>	<b>200</b>	<b>1150 (S5)</b> <b>60 (S1)</b>	<b>350/4000</b> <b>200/1500</b>	<b>This work</b>

$$^a S = (R_{\text{air}} - R_{\text{gas}})/R_{\text{air}}$$

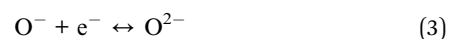
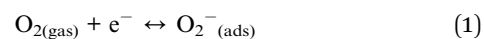
calculated by the equation  $100 \times S/|\bar{x}|$ , where  $S$  is the sample standard deviation,  $\bar{x}$  is sample mean. The RSD value of the sensor for ten pulses measurement is 92.4%, indicating the good reproducibility of the device. However, for real application, long term stability of the sensor should be studied. This work will be characterized in next step, and the data will be reported elsewhere.

For a better vision, the H<sub>2</sub>S sensing performances of the sensors based on functionalized-SnO<sub>2</sub> nanomaterials are summarized in Table 3. Compared to other results in the references, our sensor showed comparable working temperature whereas was superior in response toward much lower concentration. This means that the Ag<sub>2</sub>O decoration on the surface of SnO<sub>2</sub> NWs is suitable for development of high performance H<sub>2</sub>S gas sensor.

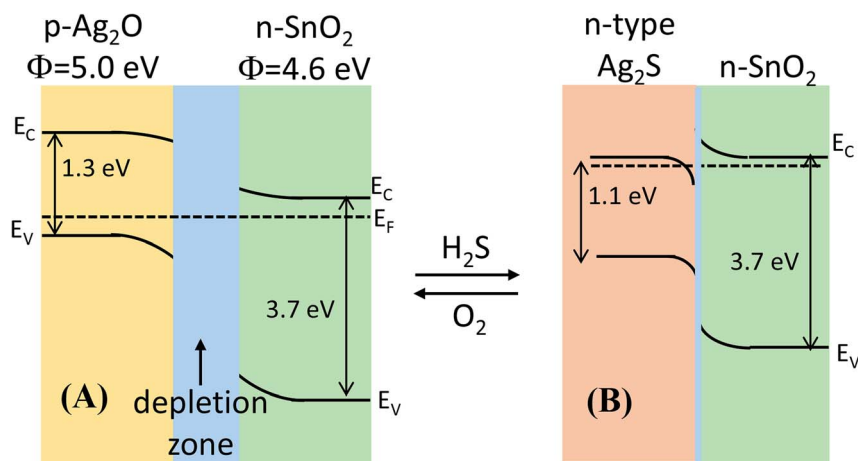
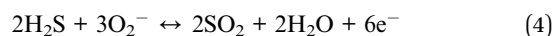
### 3.3. Gas-sensing mechanism

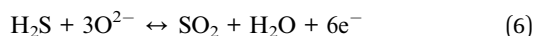
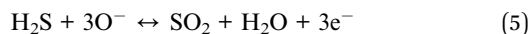
The gas-sensing mechanism of a metal oxide-based sensor is determined by the surface reaction of the analyzed gas molecule

and pre-adsorbed oxygen species.<sup>9</sup> When SnO<sub>2</sub> was exposed to air, atmospheric oxygen molecules were adsorbed on the surface of SnO<sub>2</sub> NWs to form oxygen ions (O<sub>2</sub><sup>-</sup>, O<sup>-</sup>, and O<sup>2-</sup>) by withdrawing electrons from the conduction band of SnO<sub>2</sub>, as shown in the following eqn (1)–(3):



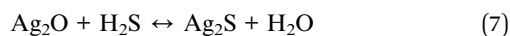
As shown in the above equations, the resistance of SnO<sub>2</sub> in air increased because of the formation of a thick conduction-depletion region. When air was replaced by H<sub>2</sub>S, the oxygen ions reacted with H<sub>2</sub>S to form SO<sub>2</sub> and H<sub>2</sub>O and then released electrons back to the conduction band, resulting in decreased SnO<sub>2</sub> resistance, as presented in eqn (4)–(6):

**Fig. 6** Energy band diagram of the formation of the p-Ag<sub>2</sub>O/n-SnO<sub>2</sub> junction in air and n-Ag<sub>2</sub>S/n-SnO<sub>2</sub> in H<sub>2</sub>S atmosphere.



However, the chemophysical processes of decoration with silver and silver oxide involved in the gas-sensing properties of metal oxides can be explained in various ways.<sup>30,52–54</sup> The mechanisms are primarily electronic and/or chemical sensitization. The electronic mechanism is related to the extension of the electron-depleted space charge region at the interface between two materials, and the latter is related to the dominance of the dissociation of gas molecules on the surface of decorated materials by spillover effect.<sup>53,54</sup>

Herein, we believed that the dissociation of gas molecules at Ag-based sites on the surface of Ag-decorated SnO<sub>2</sub> facilitated the charge-transfer reaction between sensor surface and H<sub>2</sub>S molecule. The gas-sensing mechanism of Ag<sub>2</sub>O-decorated SnO<sub>2</sub> NWs may involve the variation in band structure caused by the conversion of Ag<sub>2</sub>O into Ag<sub>2</sub>S and back to Ag<sub>2</sub>O when the test ambient switched from air to H<sub>2</sub>S and back to air, as shown in Fig. 6(A) and (B), respectively. Ag<sub>2</sub>O is a p-type narrow band-gap semiconductor (1.3 eV)<sup>47</sup> with a work function of 5.0 eV,<sup>55,56</sup> whereas SnO<sub>2</sub> is a n-type wide direct-band-gap (3.7 eV) semiconductor with a higher work function of 4.6 eV.<sup>57</sup> Given the extension of the electron-depleted region underneath Ag<sub>2</sub>O nanoparticles on the surface of SnO<sub>2</sub> NWs, the barrier at the interface between these two materials developed much more than usual.<sup>31</sup> Furthermore, the formation of a continuous series of n–p–n junctions by decorating Ag<sub>2</sub>O nanoparticles on the network of SnO<sub>2</sub> NWs, which prevented the electron current in SnO<sub>2</sub> NWs, aggravated the decrease in SnO<sub>2</sub> conductivity.<sup>21</sup> Upon exposure to H<sub>2</sub>S, Ag<sub>2</sub>O was converted into Ag<sub>2</sub>S<sup>58</sup> according to eqn (7).



The conversion of Ag<sub>2</sub>O into Ag<sub>2</sub>S occurred spontaneously because of the negative free Gibbs energy of the reaction (–224.7 kJ mol<sup>–1</sup>) at room temperature. Therefore, the conversion Ag<sub>2</sub>O into Ag<sub>2</sub>S requires less H<sub>2</sub>S gas, thus the sensor has a lower detection limit. In addition, Ag<sub>2</sub>S can be an n- or p-type semiconductor depending on its surrounding environment and the pressure.<sup>59,60</sup> The monoclinic α-Ag<sub>2</sub>S is a n-type semiconductor with a band gap of ~1.1 eV and a work function of 4.42 eV. Upon exposure to H<sub>2</sub>S, the conversion of p-type Ag<sub>2</sub>O<sup>58</sup> into n-type Ag<sub>2</sub>S destroyed the p–n junctions of Ag<sub>2</sub>O–SnO<sub>2</sub> and formed the n–n of Ag<sub>2</sub>S–SnO<sub>2</sub>, resulting in largely decreased resistance [Fig. 6(B)]. Ag<sub>2</sub>S was then re-oxidized when the sensor was in air and the p–n junctions were re-established, and the sensor resistance thus recovered to its initial value. Hence, the functionalization of silver on the surface of SnO<sub>2</sub> NWs improved their H<sub>2</sub>S-sensing properties.

## 4. Conclusion

We introduced a dip-coating method of decorating Ag<sub>2</sub>O nanoparticles on the surface of on-chip-grown SnO<sub>2</sub> NW sensors

toward H<sub>2</sub>S gas monitoring. The effect of Ag<sub>2</sub>O nanoparticles decorated on the surface of SnO<sub>2</sub> NWs on H<sub>2</sub>S gas-sensing performance was investigated. SnO<sub>2</sub> NW sensor decorated with Ag<sub>2</sub>O nanoparticles illustrated the highest response of 1150 to 1 ppm H<sub>2</sub>S at a working temperature of 200 °C with reasonable response and recovery time. Selectivity tests over high concentrations of NH<sub>3</sub> (500 ppm) and H<sub>2</sub> (500 ppm) at various working temperatures presented excellent response, selectivity, and reproducibility, demonstrating the sensor's potential application in the selective monitoring of low-level H<sub>2</sub>S gas. The high performance of the sensor was also confirmed under the light of sulfurization, which turned the band structure from p–n of Ag<sub>2</sub>O–SnO<sub>2</sub> into n–n of Ag<sub>2</sub>S–SnO<sub>2</sub>.

## Conflicts of interest

The authors hereby declare that they have no conflict of interests regarding the publication of this paper.

## Acknowledgements

This work was supported by the Vietnam National Foundation for Science and Technology Development (NAFOSTED) under grant number 103.02-2017.25.

## References

- 1 S. K. Pandey, K. H. Kim and K. T. Tang, *TrAC, Trends Anal. Chem.*, 2012, **32**, 87–99.
- 2 OSHA, *OSHA standards or regulations of Hydrogen Sulfide (H<sub>2</sub>S)*, 2005.
- 3 P. Van Tong, N. D. Hoa, H. T. Nha, N. Van Duy, C. M. Hung and N. Van Hieu, *J. Electron. Mater.*, 2018, **47**, 7170–7178.
- 4 E. Ashori, F. Nazari and F. Illas, *Int. J. Hydrogen Energy*, 2014, **39**, 6610–6619.
- 5 B. R. Doyle, *Hazardous Gases Underground: Applications to Tunnel Engineering*, Marcel Dekker, Inc., USA, 2001.
- 6 T.-T. Xu, Y.-M. Xu, X.-F. Zhang, Z.-P. Deng, L.-H. Huo and S. Gao, *Front. Chem.*, 2018, **6**, 165.
- 7 A. Mirzaei, S. S. Kim and H. W. Kim, *J. Hazard. Mater.*, 2018, **357**, 314–331.
- 8 L. Sui, T. Yu, D. Zhao, X. Cheng, X. Zhang, P. Wang, Y. Xu, S. Gao, H. Zhao, Y. Gao and L. Huo, *J. Hazard. Mater.*, 2020, **385**, 121570.
- 9 T. Seiyama, A. Kato, K. Fujiishi and M. Nagatani, *Anal. Chem.*, 1962, **34**, 1502–1503.
- 10 A. Natkazo, D. Phokharatkul, J. H. Hodak, A. Wisitsoraat and S. K. Hodak, *Sens. Actuators, B*, 2018, **260**, 571–580.
- 11 P. H. Phuoc, C. M. Hung, N. Van Toan, N. Van Duy, N. D. Hoa and N. Van Hieu, *Sens. Actuators, A*, 2020, **303**, 111722.
- 12 L. V. Thong, L. T. N. Loan and N. Van Hieu, *Sens. Actuators, B*, 2010, **150**, 112–119.
- 13 M. Batzill and U. Diebold, *Prog. Surf. Sci.*, 2005, **79**, 47–154.
- 14 N. X. Thai, N. Van Duy, C. M. Hung, H. Nguyen, T. M. Hung, N. Van Hieu and N. D. Hoa, *J. Sci. Adv. Mater. Devices*, 2020, **5**, 40–47.
- 15 S. R. Morrison, *Sens. Actuators*, 1987, **12**, 425–440.



- 16 Z. Chen, D. Pan, Z. Li, Z. Jiao, M. Wu, C.-H. Shek, C. M. L. Wu and J. K. L. Lai, *Chem. Rev.*, 2014, **114**, 7442–7486.
- 17 D. Zappa, V. Galstyan, N. Kaur, H. M. M. Munasinghe Arachchige, O. Sisman and E. Comini, *Anal. Chim. Acta*, 2018, **1039**, 1–23.
- 18 T. Maekawa, J. Tamaki, N. Miura and N. Yamazoe, *Chem. Lett.*, 1991, **20**, 575–578.
- 19 F. Shao, M. W. G. Hoffmann, J. D. Prades, R. Zamani, J. Arbiol, J. R. Morante, E. Varechkina, M. Rumyantseva, A. Gaskov, I. Giebelhaus, T. Fischer, S. Mathur and F. Hernández-Ramírez, *Sens. Actuators, B*, 2013, **181**, 130–135.
- 20 N. Van Toan, N. V. Chien, N. Van Duy, D. D. Vuong, N. H. Lam, N. D. Hoa, N. Van Hieu and N. D. Chien, *Appl. Surf. Sci.*, 2015, **324**, 280–285.
- 21 N. Van Hieu, P. Thi Hong Van, L. Tien Nhan, N. Van Duy and N. Duc Hoa, *Appl. Phys. Lett.*, 2012, **101**, 253106.
- 22 S. Zhang, P. Zhang, Y. Wang, Y. Ma, J. Zhong and X. Sun, *ACS Appl. Mater. Interfaces*, 2014, **6**, 14975–14980.
- 23 I.-S. Hwang, J.-K. Choi, S.-J. Kim, K.-Y. Dong, J.-H. Kwon, B.-K. Ju and J.-H. Lee, *Sens. Actuators, B*, 2009, **142**, 105–110.
- 24 M. N. Rumyantseva, S. A. Vladimirova, N. A. Vorobyeva, I. Giebelhaus, S. Mathur, A. S. Chizhov, N. O. Khmelevsky, A. Y. Aksechenko, V. F. Kozlovsky, O. M. Karakulina, J. Hadermann, A. M. Abakumov and A. M. Gaskov, *Sens. Actuators, B*, 2018, **255**, 564–571.
- 25 S.-W. Choi, A. Katoch, J. Zhang and S. S. Kim, *Sens. Actuators, B*, 2013, **176**, 585–591.
- 26 Y. Wang, H. Zhang and X. Sun, *Appl. Surf. Sci.*, 2016, **389**, 514–520.
- 27 M. Kaur, B. K. Dadhich, R. Singh, K. Ganapathi, T. Bagwaiya, S. Bhattacharya, A. K. Debnath, K. P. Muthe and S. C. Gadkari, *Sens. Actuators, B*, 2017, **242**, 389–403.
- 28 A. Khanna, R. Kumar and S. S. Bhatti, *Appl. Phys. Lett.*, 2003, **82**, 4388–4390.
- 29 C.-J. Chen and R.-K. Chiang, *Dalton Trans.*, 2011, **40**, 880–885.
- 30 N. Cattabiani, C. Baratto, D. Zappa, E. Comini, M. Donarelli, M. Ferroni, A. Ponzoni and G. Faglia, *J. Phys. Chem. C*, 2018, **122**, 5026–5031.
- 31 I.-S. Hwang, J.-K. Choi, H.-S. Woo, S.-J. Kim, S.-Y. Jung, T.-Y. Seong, I.-D. Kim and J.-H. Lee, *ACS Appl. Mater. Interfaces*, 2011, **3**, 3140–3145.
- 32 N. Bhardwaj and S. Mohapatra, *Ceram. Int.*, 2016, **42**, 17237–17242.
- 33 T. Yang, Q. Yang, Y. Xiao, P. Sun, Z. Wang, Y. Gao, J. Ma, Y. Sun and G. Lu, *Sens. Actuators, B*, 2016, **228**, 529–538.
- 34 P. S. Kolhe, P. M. Koinkar, N. Maiti and K. M. Sonawane, *Phys. B*, 2017, **524**, 90–96.
- 35 J. Gong, Q. Chen, M.-R. Lian, N.-C. Liu, R. G. Stevenson and F. Adami, *Sens. Actuators, B*, 2006, **114**, 32–39.
- 36 X. Wen, S. Wang, Y. Xie, X.-Y. Li and S. Yang, *J. Phys. Chem. B*, 2005, **109**, 10100–10106.
- 37 V. V. Sysoev, T. Schneider, J. Goschnick, I. Kiselev, W. Habicht, H. Hahn, E. Strelcov and A. Kolmakov, *Sens. Actuators, B*, 2009, **139**, 699–703.
- 38 T. M. Ngoc, N. Van Duy, C. M. Hung, N. D. Hoa, H. Nguyen, M. Tonezzer and N. Van Hieu, *Anal. Chim. Acta*, 2019, **1069**, 108–116.
- 39 N. Van Hieu, L. T. N. Loan, N. D. Khoang, N. T. Minh, D. T. Viet, D. C. Minh, T. Trung and N. D. Chien, *Curr. Appl. Phys.*, 2010, **10**, 636–641.
- 40 M. Tonezzer, D. T. T. Le, S. Iannotta and N. Van Hieu, *Sens. Actuators, B*, 2018, **277**, 121–128.
- 41 D. D. Trung, N. Van Toan, P. Van Tong, N. Van Duy, N. D. Hoa and N. Van Hieu, *Ceram. Int.*, 2012, **38**, 6557–6563.
- 42 N. Liang, M. Wang, L. Jin, S. Huang, W. Chen, M. Xu, Q. He, J. Zai, N. Fang and X. Qian, *ACS Appl. Mater. Interfaces*, 2014, **6**, 11698–11705.
- 43 K. H. Stern, *J. Phys. Chem. Ref. Data*, 1972, **1**, 747–772.
- 44 M. A. M. Hassan, I. R. Agoor and L. M. Raoof, *Appl. Nanosci.*, 2014, **4**, 429–447.
- 45 I.-S. Hwang, J.-K. Choi, H.-S. Woo, S.-J. Kim, S.-Y. Jung, T.-Y. Seong, I.-D. Kim and J.-H. Lee, *ACS Appl. Mater. Interfaces*, 2011, **3**, 3140–3145.
- 46 N. Van Hoang, C. M. Hung, N. D. Hoa, N. Van Duy, N. Van Toan, H. S. Hong, P. T. Hong Van, N. T. Sơn, S.-G. Yoon and N. Van Hieu, *J. Alloys Compd.*, 2020, **826**, 154169.
- 47 Y. Zhao, C. Tao, G. Xiao and H. Su, *RSC Adv.*, 2017, **11**, 11211–11221.
- 48 D. J. Late, Y.-K. Huang, B. Liu, J. Acharya, S. N. Shirodkar, J. Luo, A. Yan, D. Charles, U. V. Waghmare, V. P. Dravid and C. N. R. Rao, *ACS Nano*, 2013, **7**, 4879–4891.
- 49 P. K. Kannan, D. J. Late, H. Morgan and C. S. Rout, *Nanoscale*, 2015, **7**, 13293–13312.
- 50 K.-I. Choi, H.-J. Kim, Y. C. Kang and J.-H. Lee, *Sens. Actuators, B*, 2014, **194**, 371–376.
- 51 A. Stanoiu, S. Somacescu, C. E. Simion, J. M. Calderon-Moreno and O. G. Florea, in *2017 International Semiconductor Conference (CAS)*, IEEE, 2017, pp. 93–96.
- 52 X. Chen, Z. Guo, W.-H. Xu, H.-B. Yao, M.-Q. Li, J.-H. Liu, X.-J. Huang and S.-H. Yu, *Adv. Funct. Mater.*, 2011, **21**, 2049–2056.
- 53 S. Matsushima, Y. Teraoka, N. Miura and N. Yamazoe, *Jpn. J. Appl. Phys.*, 1988, **27**, 1798–1802.
- 54 N. Yamazoe, Y. Kurokawa and T. Seiyama, *Sens. Actuators*, 1983, **4**, 283–289.
- 55 D. Sarkar, C. K. Ghosh, S. Mukherjee and K. K. Chattopadhyay, *ACS Appl. Mater. Interfaces*, 2013, **5**, 331–337.
- 56 H. Won Choi, S. Young Kim, K.-B. Kim, Y.-H. Tak and J.-L. Lee, *Appl. Phys. Lett.*, 2005, **86**, 12104.
- 57 V. Van Quang, N. Van Dung, N. Sy Trong, N. Duc Hoa, N. Van Duy and N. Van Hieu, *Appl. Phys. Lett.*, 2014, **105**, 13107.
- 58 L.-M. Lyu and M. H. Huang, *Chem.-Asian J.*, 2013, **8**, 1847–1853.
- 59 J. Zhang, C. Liu, X. Zhang, F. Ke, Y. Han, G. Peng, Y. Ma and C. Gao, *Appl. Phys. Lett.*, 2013, **103**, 82116.
- 60 J. Jang, K. Cho, S. H. Lee and S. Kim, *Mater. Lett.*, 2008, **62**, 1438–1440.

