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A highly stretchable strain sensor based on CNT/graphene/fullerene-SEBS

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Recently, highly stretchable strain sensors have attracted considerable attention. Identifying alternatives to sensitive unit materials and flexible substrates is critical in the fabrication of sensors. Herein, a ternary hybrid carbon material consisting of carbon nanotubes (CNTs), graphene, and fullerene was chosen due to its dense interconnections and robust mechanism. Additionally, the cost-effective fabrication of styrene ethylene butylene styrene (SEBS) provides a platform for the strong adhesion of substrates, which contributes to the strong interaction between the substrates and the sensitive unit materials. Furthermore, the intrinsically high elasticity of SEBS allows the sensors to endure large stretching ranges. Owing to the above-mentioned merits, the fabricated sensor based on CNT/graphene/fullerene-SEBS has a high conductivity of 5.179 S m^{-1} , a moderate gauge factor (GF) of 15, an optimum stretching range of 203%, a linearity of 136% ($R^2 = 0.998$), and adaptive-rate repeatability, which reveals its potential in the fields of human motion monitoring and scalable applications.

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1 Introduction

With the development of smart technology, flexible electronic devices have been applied in various fields, including health monitoring,¹ human motion,² flexible photoelectrics,³ artificial muscle design,⁴ and intelligent weaving.⁵ Due to better stretchability,^{6,7} flexibility,^{8,9} skin adaptability and sensitivity of flexible electronic devices,^{10–13} much attention has been given to the potential applications of highly stretchable, linear,^{14,15} and electrically conductive strain sensors. In particular, flexible substrates with wide stretching ranges^{16–18} and hybrid sensitive unit nanomaterials are still hot topics in scientific research.

In general, flexible sensors are mainly composed of two parts, sensitive unit materials and flexible substrates. Traditionally, carbon/metal nanomaterials and polydimethylsiloxane (PDMS), as common sensitive unit materials and flexible substrates, respectively, have been widely applied in flexible strain sensors. Metal nanomaterials were extensively studied due to their excellent electrical conductivity and gauge factor (GF).^{19–22} Amjadi *et al.* reported a highly stretchable and sensitive strain sensor based on Ag nanowire (AgNW)/PDMS with a stretchability of around 70% and a GF of ≈ 14 .²³ Kim *et al.* introduced a type of highly sensitive stretchable strain sensor based on Ag flakes/Ag nanoparticles (AgNPs), which reached a strain range of 80% and a GF of 7.1.²⁴ Additionally, carbon

nanomaterials were highly reported because of their outstanding consistency, excellent mechanical properties, and good electrical conductivity.^{25–28} CNT, a typical delegate of the 1-dimensional (1-D) carbon nanomaterial family, can withstand a wide range of stretching without causing fracture of the interconnections due to its high aspect ratio. Due to this characteristic, 1-D carbon nanomaterials also show a low relative resistance when stretched. Graphene, a 2-dimensional (2-D) carbon material, has a low aspect ratio, so it also performed a relatively large relative resistance and easy disconnection. Because of this, there has always been a contradiction between stretchability and gauge factor (GF) in CNT–PDMS and graphene–PDMS based sensors. For example, Yamada *et al.* studied a stretchable strain sensor based on CNTs, which exhibited high consistency and endured a 280% strain range with a dramatically low GF of around 0.82.²⁹ Jeong *et al.* reported highly stretchable and sensitive strain sensors based on graphene with a moderate stretchability of 70% and a high GF of up to 29.³⁰ To balance the characteristics of the two material systems, it was feasible to combine CNTs with graphene to resolve the contradiction between stretchability and GF. Cai *et al.* reported that under the synergistic mechanism of CNTs and graphene foam, experimental results indicated that the sensor had a large stretching range (85%) and high gauge factor ($\text{GF} \approx 20.5$).³¹ In order to further defer the crack propagation of the sensitive unit materials and ensure the normal operation of the sensor under huge enforced strain, fullerene, as a 0-dimensional (0-D) carbon material, was used to efficiently hinder the breakage and lubricate the adjacent sliding of the 2-D graphene layers.³² This inspired us to offset the insufficient stretching range. Due to the

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moderate stretching ranges of PDMS, much effort was focused on adopting other substrates including thermoplastic urethane (TPU) and natural rubber.^{33,34} However, complex preparation processes involving chemical reagents are in opposition to environmentally friendly considerations. Thus, tolerable-strain merits and easy preparation protocols are needed to meet the demand for substrates. Zunfeng *et al.* took advantage of the high elasticity of SEBS to fabricate superelastic electronic sensors,³⁵ which provided us with research ideas for manufacturing highly stretchable sensors (Table 1).

Herein, based on the excellent consistency of CNTs, good conductivity of graphene, and the spherical lubrication of fullerene, trinary hybrid carbon nanomaterials CNT/graphene/fullerene were selected as sensitive unit materials in order to provide excellent conductivity and large strain ranges. SEBS-based flexible substrates were fabricated by mixing and heating liquid paraffin with fixed quantities of SEBS powder, which proved to be an easy and cost-effective fabrication method. The substrate fabricated from the hybrid SEBS powder and liquid paraffin possessed strong adhesion properties, which gave rise to a strong interaction with sensitive unit materials. As a result, inherent adhesive merit was proved by sticking to various materials and tape testing. By comparing the electrical performance of single CNT-SEBS based sensors, binary CNT/graphene-SEBS based sensors and the trinary CNT/graphene/fullerene-SEBS based sensor, we concluded that the trinary CNT/graphene/fullerene-SEBS based sensor had the highest conductivity and stretching and linearity ranges but a relatively moderate gauge factor (GF). Moreover, cyclic tests with different rates and stretching ranges were performed to demonstrate stable repeatability and reproduction. Finally, human monitoring measurements could be performed in real time while blowing and bending wrists and fingers, indicating the potential for future applications.

2 Experiment

2.1 Fabrication of flexible strain sensors

A schematic of the sensor is shown in Fig. 1(a). First, the SEBS (TSRC Nantong Industries Co., Ltd.) powder and liquid paraffin (Shangqiu Liangfeng Health Products Co., Ltd.) were mixed in a certain mass ratio (1 : 3 in this experiment) and stirred in a beaker to obtain a uniform mixture. A certain amount of the

mixture was placed into a vector with a length, width, and height of $60 \times 30 \times 20 \text{ mm}^3$, respectively. The ark vector was placed in a high temperature resistance furnace (Shanghai Boxun Industrial Co., Ltd. Medical Equipment Factory) and heated to $220 \text{ }^\circ\text{C}$ for 30 minutes. Finally, it was placed at room temperature until the molten SEBS cooled, solidified, and peeled off from the porcelain ark vector. Different proportions of the sensitive unit materials (1 : 1 for CNT/graphene and 1 : 1 : 0.23 for CNT/graphene/fullerene) dissolved in 30 mL of deionized water were placed in an ultrasonic cell pulverizer (Branson ultrasonic Shanghai Co., Ltd.) for 40 min at 40% power. CNT was purchased from Chengdu Organic Chemistry Co., Ltd., the Chinese Academy of Sciences. Graphene was purchased from Shenzhen Guoheng Technology Co., Ltd and fullerene was purchased from Shanghai Macklin Biochemical Co., Ltd. Eventually, the fabricated materials were evaporated to dryness in air and ground into a powder. The prepared SEBS substrate was spread on a Petri dish and double transparent tape stripes were placed symmetrically on both sides of the long axis of the substrate leaving a width of 5 mm in the middle. Then, two conductive tapes were symmetrically attached along both ends of the long axis with a length of 18 mm reserved in the middle. Next, the sensitive unit materials were homogeneously filled in the middle groove of around 0.25 mm in thickness. Eventually, the sensor fabrication was completed after the transparent tape stripes on both sides were torn off. Fig. 1(b) represents the finished SEBS substrate with an approximate length, width, and height of $60 \times 30 \times 2 \text{ mm}^3$. Fig. 1(c), (d), and (e) represent the state of the sensor under initial, bending, and stretching conditions, respectively.

2.2 Characterization and properties testing of sensors

The topography of the sensitive unit materials and substrate were characterized by scanning electron microscopy (SEM Hitachi Corporation, Tokyo, Japan) employed at 15 kV and transmission electron microscopy (TEM JEOL Corporation, Tokyo, Japan) under 200 kV. A tensile test was performed on a flexible substrate prepared by mixing different SEBS powders and paraffin oil. Primarily, a digital display force gauge was separately fixed on a self-assembled optical displacement platform to test the relationship between stress and strain. Then, the probes of a digital source meter (Keithley 2400) were clamped onto the sensor to measure its I - V curve, conductivity, and

Table 1 Summary of performance compared to previously reported flexible strain sensors

Sensitive unit materials	Substrate	Strain	Max linearity	GF	Reference
AgNW	PDMS	70%	Up to 50%	2–14	23
Ag fakes/AgNPs	PDMS	80%	N/A	7.1	24
CNT	PDMS	280%	Up to 140%	0.82	29
Graphene	PDMS	70%	N/A	15–29	30
CNT/graphene	PDMS	85%	N/A	20.5	31
AgNW/graphene/fullerene	Polyurethane	62%	32%	2392.9	32
Graphene	TPU	30%	N/A	0.78–17.7	33
CNT	Natural rubber	100%	N/A	43.5	34
CNT	SEBS	700–1320%	N/A	0.005	35
CNT/graphene/fullerene	SEBS	203%	Up to 136%	15	This work



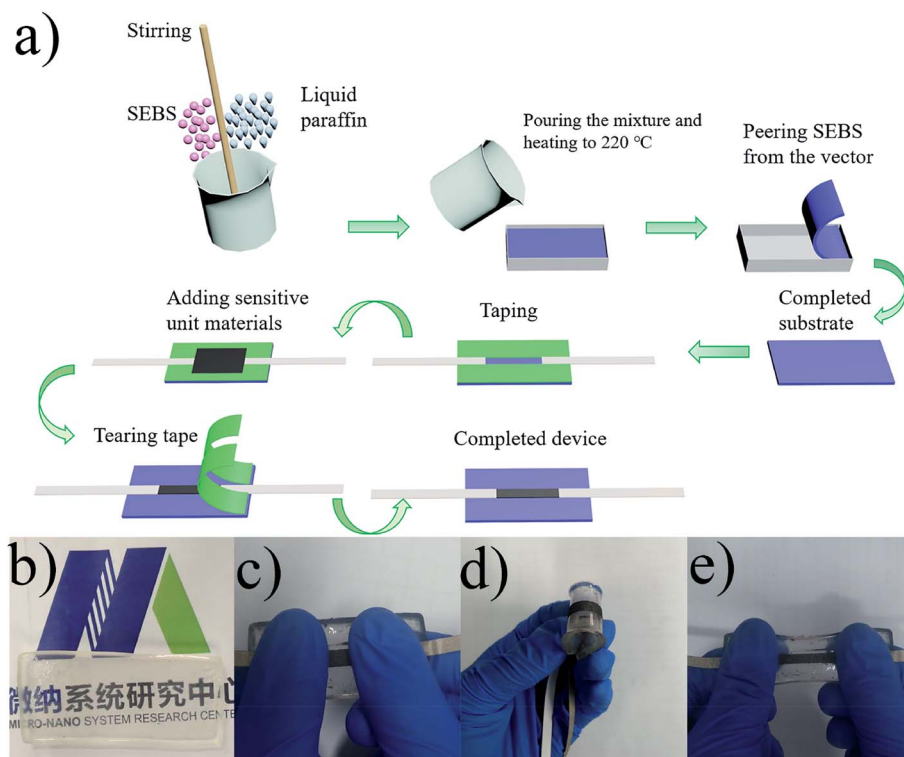


Fig. 1 (a) Schematic of the CNT/graphene/fullerene-SEBS based sensor. (b) Finished SEBS substrate film. (c) Initial state of the sensor. (d) Bending state of the sensor. (e) Stretching state of the sensor.

relative resistance to strain. Additionally, the sensor was mounted on an electric cycle motor for cyclic testing at different stretching lengths and speeds. Finally, blowing, finger bending, and wrist bending detection was tested.

3 Results and discussion

3.1 Characterization of the sensor

The geometry of the hybrid CNT/graphene/fullerene is exhibited in Fig. 2(a-c). The topography of CNTs and fullerene under TEM could be clearly observed. The size of the tubular CNTs was around $10 \mu\text{m} \times 10 \text{nm}$. The diameter of the spherical fullerene was 5 nm. The blue circle in Fig. 2(c) presents a SEM image (under 2500 times magnification) of the lamellar structure of graphene, where the graphene has a sheet diameter of approximately $12 \mu\text{m}$. Fig. 2(d) represents a cross-section of SEBS and the sensitive unit materials. Fig. 2(a-c) demonstrate the composition and geometric topography of the sensitive unit materials. Fig. 2(d) displays a strong interaction between SEBS and the sensitive unit materials, which arises from the inherently adhesive surface of SEBS.

3.2 Performance test of the substrate

We then explored the effect of different ratios of SEBS powder and liquid paraffin on the mechanical properties of the prepared SEBS substrate. A digital display force gauge was fixed on the self-assembled optical displacement platform to understand the relation between stress and strain in the substrates with different

mixing ratios. For a certain proportion of the substrates shown in Fig. 3, as the applied strain increased the stress ascended correspondingly, which revealed a directly proportional

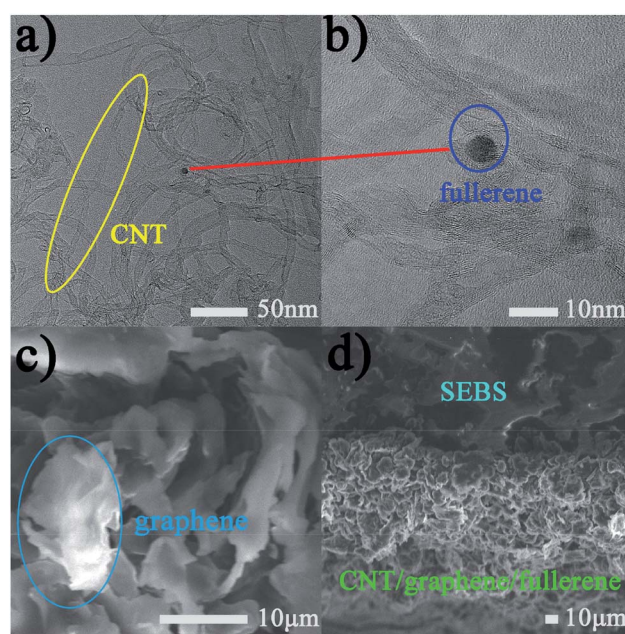


Fig. 2 TEM images showing (a) CNT and (b) fullerene in the hybrid CNT/graphene/fullerene unit. SEM images showing (c) graphene in the hybrid CNT/graphene/fullerene unit. (d) Cross-section of SEBS and hybrid CNT/graphene/fullerene.



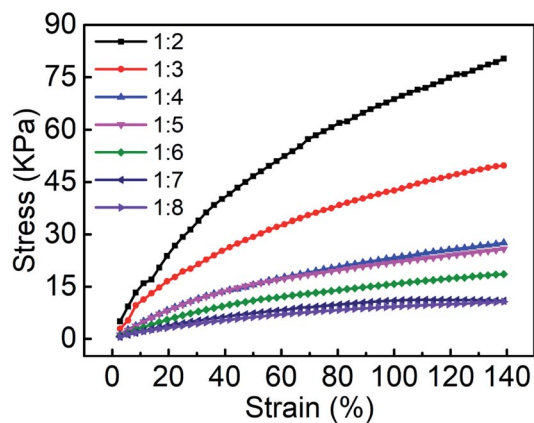


Fig. 3 Stress–strain curves of substrates of different ratios.

relationship. Secondly, under the same strain conditions, the stress gradually decreased as the liquid paraffin mass percentage increased. In summary, the toughness of the substrate produced reduced due to the loss of SEBS powder. In order to endow the substrate with a larger stretching range and better toughness, the 1 : 3 blend ratio of the substrates was chosen for this experiment to ensure both transparency and consistency.

The flexible substrate demonstrated good adhesion to different materials. As shown in Fig. 4 (a–d), the substrate could adhere to a plastic centrifuge tube, polylactic acid, an iron long tail clip, and a glass beaker, respectively. Basically, the ideal adhesion properties promoted this combination of sensitive unit materials and flexible substrates. In order to compare the adhesion of the SEBS substrate to a traditional substrate, tape tests were utilized. The adhesion of SEBS and PDMS to sensitive unit materials was studied. SEBS and PDMS with the same length, width, and height of 60 mm, 30 mm, and 2 mm were used as

samples in the experiment. The black and red curves in Fig. 4(e) represent the relative resistance at different tear times of the PDMS-based and SEBS-based sensors. It can be observed that tearing removed the sensitive unit materials from PDMS after 3 attempts and the relative resistance suddenly ascended to the open state. Comparatively, stable relative resistance could be maintained after tearing 11 times on SEBS. This is attributed to the excellent adhesive properties of the sensitive unit materials, which could comply with the deformation of the substrates. This great protective mechanism mitigates the impact of external environmental factors for practical applications.

3.3 Performance test of sensors

The voltage–current (I – V) characteristics of the conductivity of the sensors based on CNTs, CNT/graphene, and CNT/graphene/fullerene can be observed in Fig. 5(a–c). Within the voltage range of -1000 mV to 1000 mV, current and voltage showed a perfectly linear dependence and favorable resistance characteristics. Conductivity (σ) can be calculated by the formula $\sigma = L/RS$, in which L , R , and S represent the length, resistance, and cross-sectional area of the sensitive unit materials, respectively. The conductivities of the sensors in the initial state were calculated from the data to be 1.684 S m^{-1} , 2.932 S m^{-1} and 5.179 S m^{-1} , as displayed in Fig. 5(d). By comparing the conductivity of the CNT and CNT/graphene-based sensors, we evidently found that the CNT/graphene-based sensor has a higher conductivity than the single CNT-based sensor, which proves that the addition of graphene boosts the conductivity of the materials. Notably, fullerenes, as 0-D spherical materials, have good lubricating properties and contribute to the reduction of crack propagation in sensitive unit materials and act as interlayer lubrication to bridge CNT and graphene units. Hence, fullerene materials were added to the CNT/graphene hybrid materials. It was found that the conductivity of the sensor based on CNT/graphene/fullerene

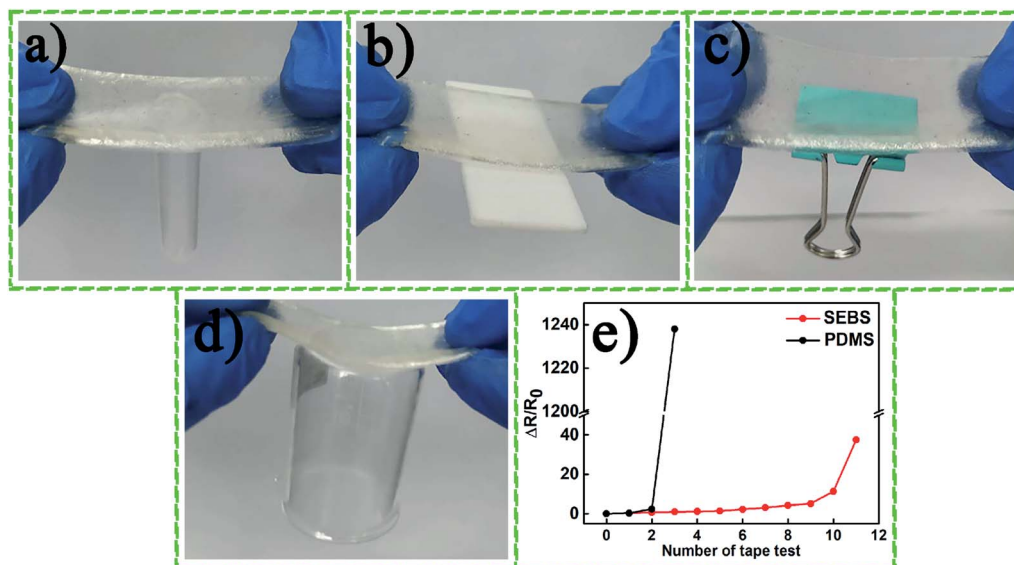


Fig. 4 The SEBS substrate film adhered to (a) a plastic centrifuge tube, (b) polylactic acid, (c) an iron long tail clip, and (d) a glass beaker. (e) Tape test results.



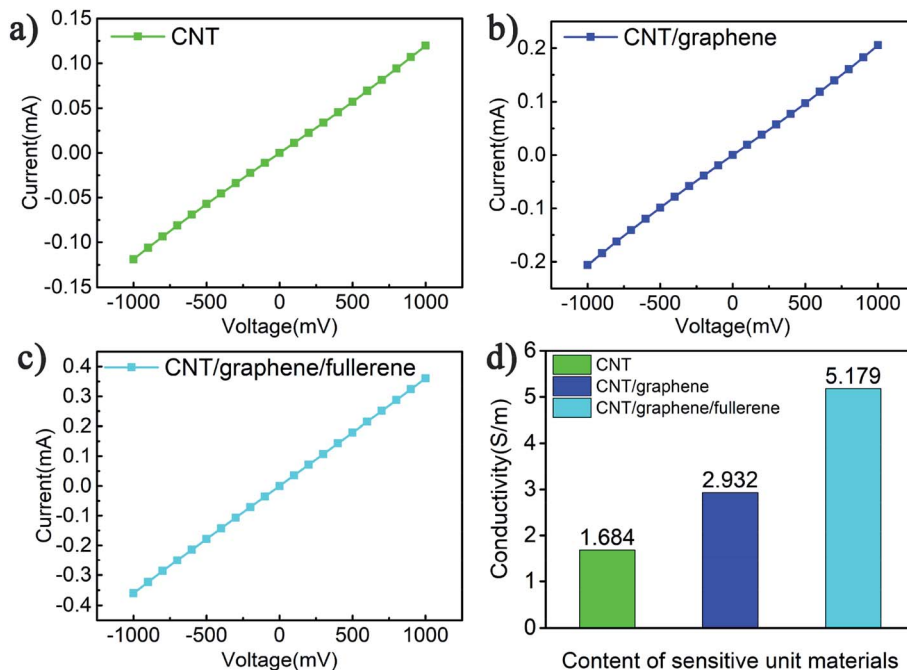


Fig. 5 I - V characteristic curve of the (a) CNT-based sensor, (b) CNT/graphene-based sensor, and (c) CNT/graphene/fullerene-based sensor. (d) Histogram of conductivity.

materials was higher than that of the sensor based on CNT/graphene. The results demonstrated that fullerene acting as an electrical bridge endowed CNTs and graphene with better conductive paths, effectively enhancing the conductivity.

Next, the relative resistance of the three sensors was researched under relative strain ranges. The gauge factor (GF) can be calculated using the equation

$$GF = \frac{\Delta R/R_0}{\varepsilon}$$

where, $\Delta R/R_0$ and ε represent the relative resistance and strain range, respectively. As shown in Fig. 6(a), the CNT-based sensor has a strain range of 120%, an optimized linearity of approximately 100% ($R^2 = 0.994$), and a maximum relative resistance of approximately 100, which was calculated to obtain a gauge factor (GF) of 86, as seen in Fig. 6(d). In Fig. 6(b), the maximum strain range of the sensor based on CNT/graphene was 120% with a linearity of approximately 50% ($R^2 = 0.971$). There was no significant maximum strain change compared with the former system. In contrast, the maximum relative resistance sharply rose to more than 136. The gauge factor (GF) of the CNT/graphene-based sensor obtained by calculation was 148 significantly better than the former attributed to smaller initial resistance, as well as mechanism prone to crack propagation and interlayer sliding of graphene. Finally, based on the CNT/graphene/fullerene sensor shown in Fig. 6(c), the addition of fullerene contributed to the dense interconnections between the sensitive unit materials. As a result, the maximum strain range was significantly increased to 203% and the gauge factor (GF) dropped significantly to 15. Fortunately, the CNT/

graphene/fullerene-based sensor maintained outstanding linearity of up to 136% ($R^2 = 0.998$).

The principle of structural transformation of the sensitive unit materials during the stretching process was revealed in Fig. 7. In the initial state, the three sensitive unit materials were closely distributed. In the CNT-based sensor, the conductive cross-links mainly existed in the contact resistance and the tunneling resistance between the CNT units. In the CNT/graphene-based sensor, the connecting resistance between CNT and graphene served as an essential electrical bridge. Similarly, in the sensor based on CNT/graphene/fullerene, the interconnections are primarily present in the bonding of CNTs, graphene, and fullerene. In the case of a strain range of 60%, the cross-links in the CNT-based sensor between the CNT units were partially broken, causing a surge in the resistance. Similarly, the connection resistance between the CNT units and graphene and the layer resistance of the graphene were vulnerable to disruption. Notably, the sheet-like structure of graphene is more susceptible to crack propagation. Therefore, the relative resistance of the CNT/graphene sensor under the same strain conditions was higher than that of the CNT-based sensor. In the end, due to fullerene's inherent lubricity, it was possible to effectively reduce interlaminar friction and crack generation in graphene and ensure a conductive path between CNT and CNT/graphene, resulting in the lowest rate of change than previous sensors. Under the 120% strain condition, the connection resistance between the CNT-based and CNT/graphene-based sensors was almost in the critical state of fracture and the relative resistance reached a maximum. In comparison, the sensor based on CNT/graphene/fullerene possessed a large stretching range and a low resistance



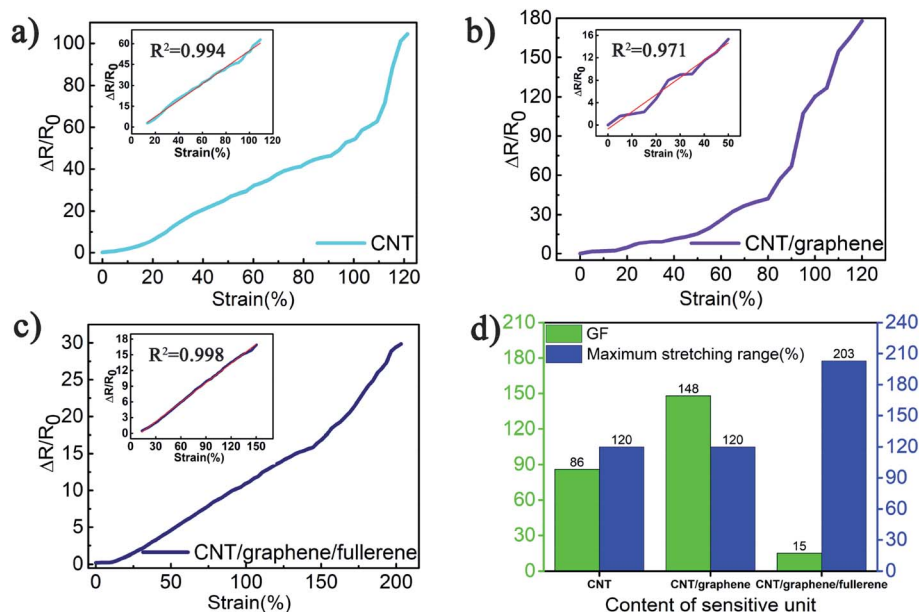


Fig. 6 Strain-resistance curve of the (a) CNT-based sensor, (b) CNT/graphene-based sensor, and (c) CNT/graphene/fullerene-based sensor. (d) GF and maximum strain range histogram corresponding to sensors of different sensitive unit materials.

change rate because of interlayer lubrication and the anti-cracking mechanism of fullerene.

In order to investigate the stability of relative resistance based on the CNT/graphene/fullerene sensor under different strain ranges and tensile rates, the sensors were mounted on an electric cycle motor and the results were displayed in real time on a digital source meter. As shown in Fig. 8(a), the strain range was set to 60% and the sensor was uniformly stretched at a rate of 2 mm s^{-1} , 4 mm s^{-1} , and 6 mm s^{-1} . The results show the resistance changes during the cyclic tensile test at three different rates, and a very stable maximum relative resistance of about 2.5 was recorded. Similarly, the sensor in Fig. 8(b) was tested at a rate of 2 mm s^{-1} , 4 mm s^{-1} , and 6 mm s^{-1} under 100% strain and the measurements showed that the maximum

relative resistance was approximately 7. For long-time cyclic tests, the sensors were subjected to 60% and 100% strain ranges at a 6 mm s^{-1} stretching rate. As displayed in Fig. 8(c and d), the properties were maintained even after a number of repetitive cycles (187 and 112 within 300 seconds). As can be seen from the results, the maximum relative resistance was basically constant under long-term repetition.

In order to achieve real-time human monitoring of movement with the sensors, we conducted three sets of application tests. Firstly, the two ends of the sensor were fixed on two horizontal displacement platforms and blowing was performed against the center of the sensor. Due to the bending angle (θ_a) generated by blowing on the sensor (Fig. 9(a)), the sensitive unit material was slightly deformed and the measured maximum

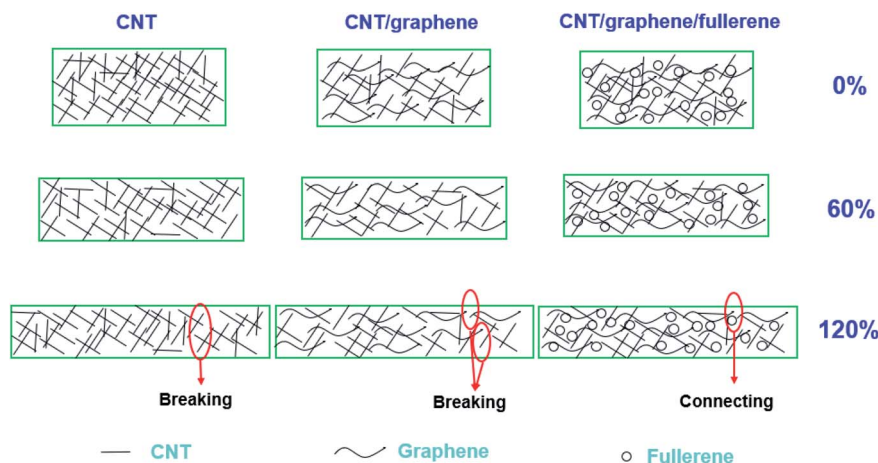


Fig. 7 Schematic diagram of the variation principle of the CNT-based sensor, CNT/graphene-based sensor, and the CNT/graphene/fullerene-based sensor under a strain of 0, 60%, and 120%.



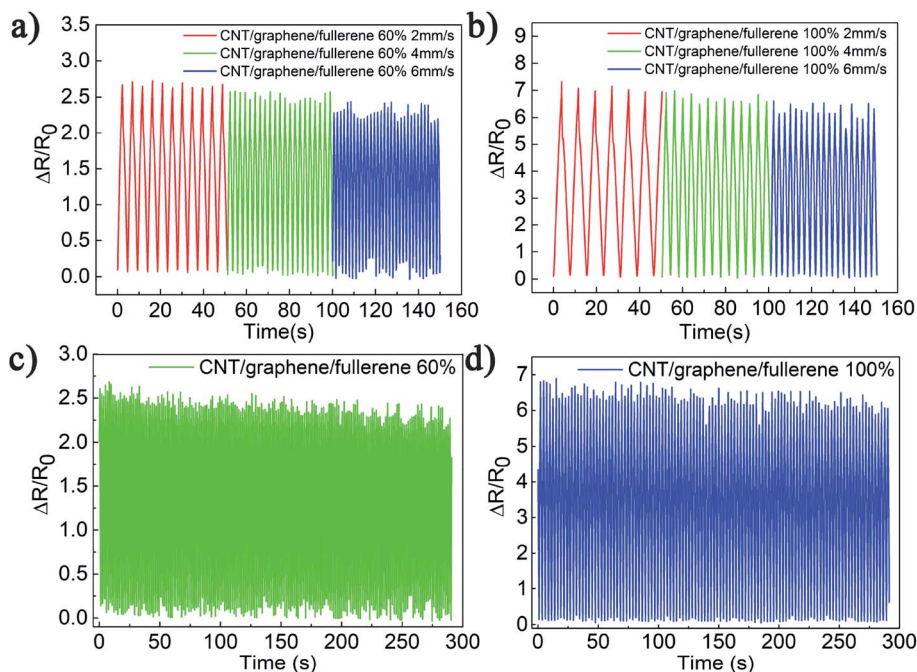


Fig. 8 Curve of resistance change rate with time at three different rates (2 mm s^{-1} , 4 mm s^{-1} , and 6 mm s^{-1}) under (a) 60% and (b) 100% strain conditions. Long-time cyclic tests at (c) 60% and (d) 100% strain conditions at a 6 mm s^{-1} stretching rate.

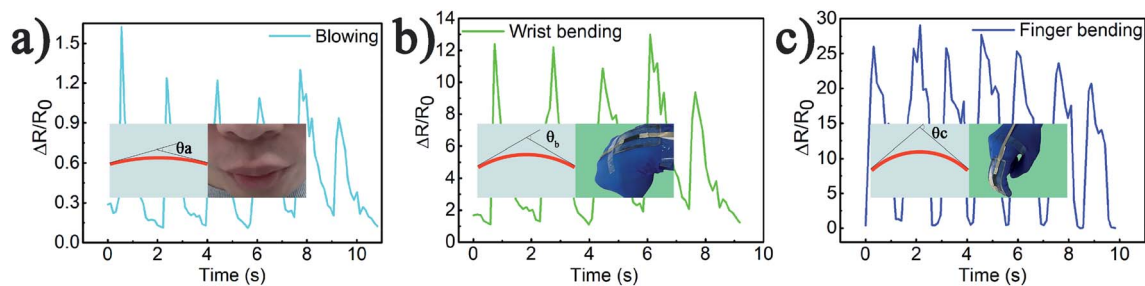


Fig. 9 Human monitoring applications: (a) blowing air, (b) wrist bending, and (c) finger bending.

resistance change rate was approximately 1.4. Secondly, due to the increased curvature (θ_b) produced by the bending of the wrist in Fig. 9(b), many cracks were observed in the sensitive unit materials, resulting in a larger relative resistance of around 12. Finally, a finger bending test was performed since the finger has the most flexible range of motion. As seen in Fig. 9(c), the bending angle (θ_c) was the largest due to huge crack propagation and the largest relative resistance of approximately 25 was observed. Therefore, it was shown that the sensor based on CNT/graphene/fullerene-SEBS can effectively monitor human movement in real time, which proves the practical value of this research for future strain sensor technologies.

4 Conclusion

In this paper, CNT/graphene/fullerene-based sensors were employed as flexible strain sensors and were fabricated from different sensitive unit materials based on SEBS. First, we performed SEM and TEM characterization of the sensor to observe

the micro appearance and geometric architecture of the sensitive unit materials and substrates, which confirmed the composition of the sensitive unit material and the close combination of the sensitive unit materials with the substrate. Then we conducted performance tests on the substrate. The mechanical properties of the flexible substrates were explored by mixing different SEBS powders with liquid paraffin and a ratio of 1 : 3 was selected as the substrate for further experiments. Through the adhesive experiment, the SEBS-based substrate was shown to demonstrate a perfect adhesion effect, which was advantageous for making sensitive unit materials conform to deformation inseparably. Finally, an electrical performance test was performed on the sensor. *I-V* characteristics and strain-resistance regulation of the sensitive unit materials based on CNTs, CNT/graphene, and CNT/graphene/fullerene were separately discussed to explore the impact of the CNT/graphene/fullerene synergetic mechanism. It was shown that the CNT/graphene/fullerene-SEBS based sensor



maintained an ideal conductivity of 5.179 S m^{-1} and a maximum sensing range of up to 203% with a fitted linearity of 136% ($R^2 = 0.998$). The comparison experiment confirmed that the sensors based on CNT/graphene/fullerene could endure long-term cyclic tests and retain stable rate-independent repeatability and reproduction. In order to realize the practical applications of the sensor, three human monitoring applications causing different bending curvatures were tested separately, which is expected to inspire further research in the field of human motion monitoring.

Conflicts of interest

The authors declare there are no conflicts of interest.

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References

- 1 Q. Hua, J. Sun, H. Liu, R. Bao, R. Yu, J. Zhai, C. Pan and Z. L. Wang, *Nat. Commun.*, 2018, **9**, 244.
- 2 G. Cai, J. Wang, K. Qian, J. Chen, S. Li and P. S. Lee, *Adv. Sci.*, 2017, **4**, 1600190.
- 3 H. Li, D. Yang, T. Zhang, P. Zhang, F. Wang, C. Qin, R. Yang, Z. D. Chen and S. Li, *J. Mater. Sci.*, 2019, **54**, 11556–11563.
- 4 D. J. Roach, C. Yuan, X. Kuang, V. C.-F. Li, P. Blake, M. L. Romero, I. Hammel, K. Yu and H. J. Qi, *ACS Appl. Mater. Interfaces*, 2019, **11**, 19514–19521.
- 5 M. Zhang, C. Wang, H. Wang, M. Jian, X. Hao and Y. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1604795.
- 6 S. Wang, J. Xu, W. Wang, G.-J. N. Wang, R. Rastak, F. Molina-Lopez, J. W. Chung, S. Niu, V. R. Feig and J. Lopez, *Nature*, 2018, **555**, 83.
- 7 J. Gu, D. Kwon, J. Ahn and I. Park, *ACS Appl. Mater. Interfaces*, 2020, **12**, 10908–10917.
- 8 Q. Meng, Z. Liu, S. Han, L. Xu, S. Araby, R. Cai, Y. Zhao, S. Lu and T. Liu, *J. Mater. Sci.*, 2019, **54**, 10856–10870.
- 9 S. Zhao, J. Li, D. Cao, G. Zhang, J. Li, K. Li, Y. Yang, W. Wang, Y. Jin and R. Sun, *ACS Appl. Mater. Interfaces*, 2017, **9**, 12147–12164.
- 10 D. Son, J. Kang, O. Vardoulis, Y. Kim, N. Matsuhisa, J. Y. Oh, J. W. To, J. Mun, T. Katsumata and Y. Liu, *Nat. Nanotechnol.*, 2018, **13**, 1057.
- 11 K. Zhao, W. Niu and S. Zhang, *J. Mater. Sci.*, 2020, **55**, 2439–2453.
- 12 C. Shao, M. Wang, L. Meng, H. Chang, B. Wang, F. Xu, J. Yang and P. Wan, *Chem. Mater.*, 2018, **30**, 3110–3121.
- 13 L. Li, Y. Bai, L. Li, S. Wang and T. Zhang, *Adv. Mater.*, 2017, **29**, 1702517.
- 14 Z. Wang, J. Chen, Y. Cong, H. Zhang, T. Xu, L. Nie and J. Fu, *Chem. Mater.*, 2018, **30**, 8062–8069.
- 15 X. Li, H. Hu, T. Hua, B. Xu and S. Jiang, *Nano Res.*, 2018, **11**, 5799–5811.
- 16 Y. Zheng, Y. Li, K. Dai, Y. Wang, G. Zheng, C. Liu and C. Shen, *Compos. Sci. Technol.*, 2018, **156**, 276–286.
- 17 J. Sun, Y. Zhao, Z. Yang, J. Shen, E. Cabrera, M. J. Lertola, W. Yang, D. Zhang, A. Benatar and J. M. Castro, *Nanotechnology*, 2018, **29**, 355304.
- 18 L. Wang, J. Luo, Y. Chen, L. Lin, X. Huang, H. Xue and J. Gao, *ACS Appl. Mater. Interfaces*, 2019, **11**, 17774–17783.
- 19 E. J. Markvicka, M. D. Bartlett, X. Huang and C. Majidi, *Nat. Mater.*, 2018, **17**, 618.
- 20 H. Liu, Q. Li, S. Zhang, R. Yin, X. Liu, Y. He, K. Dai, C. Shan, J. Guo and C. Liu, *J. Mater. Chem. C*, 2018, **6**, 12121–12141.
- 21 W. Gao, H. Ota, D. Kiriya, K. Takei and A. Javey, *Acc. Chem. Res.*, 2019, **52**, 523–533.
- 22 K. K. Kim, S. Hong, H. M. Cho, J. Lee, Y. D. Suh, J. Ham and S. H. Ko, *Nano Lett.*, 2015, **15**, 5240–5247.
- 23 M. Amjadi, A. Pichitpajongkit, S. Lee, S. Ryu and I. Park, *ACS Nano*, 2014, **8**, 5154–5163.
- 24 I. Kim, K. Woo, Z. Zhong, P. Ko, Y. Jang, M. Jung, J. Jo, S. Kwon, S.-H. Lee and S. Lee, *Nanoscale*, 2018, **10**, 7890–7897.
- 25 J. J. Park, W. J. Hyun, S. C. Mun, Y. T. Park and O. O. Park, *ACS Appl. Mater. Interfaces*, 2015, **7**, 6317–6324.
- 26 F.-Z. Zheng, Z.-Y. Zhou, X. Yang, Y.-K. Tang and Y. Wu, *IEEE Trans. Nanotechnol.*, 2010, **10**, 694–698.
- 27 C. Xu, S. Hu, R. Zhang, H. Hu, C. Ying, F. Zhang, Q. Liu and X. Fu, *Polym. Bull.*, 2019, 1–15.
- 28 P.-N. Song and J.-L. Hong, *J. Mater. Chem. C*, 2019, **7**, 13161–13175.
- 29 T. Yamada, Y. Hayamizu, Y. Yamamoto, Y. Yomogida, A. Izadi-Najafabadi, D. N. Futaba and K. Hata, *Nat. Nanotechnol.*, 2011, **6**, 296.
- 30 Y. R. Jeong, H. Park, S. W. Jin, S. Y. Hong, S. S. Lee and J. S. Ha, *Adv. Funct. Mater.*, 2015, **25**, 4228–4236.
- 31 Y. Cai, J. Shen, Z. Dai, X. Zang, Q. Dong, G. Guan, L. J. Li, W. Huang and X. Dong, *Adv. Mater.*, 2017, **29**, 1606411.
- 32 X. Shi, S. Liu, Y. Sun, J. Liang and Y. Chen, *Adv. Funct. Mater.*, 2018, **28**, 1800850.
- 33 H. Liu, Y. Li, K. Dai, G. Zheng, C. Liu, C. Shen, X. Yan, J. Guo and Z. Guo, *J. Mater. Chem. C*, 2016, **4**, 157–166.
- 34 S. Wang, X. Zhang, X. Wu and C. Lu, *Soft Matter*, 2016, **12**, 845–852.
- 35 Z. Liu, S. Fang, F. A. Moura, J. Ding, N. Jiang, J. Di, M. Zhang, X. Lepró, D. Galvao and C. Haines, *Science*, 2015, **349**, 400–404.

