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# Organic template-based ZnO embedded Mn<sub>3</sub>O<sub>4</sub> nanoparticles: synthesis and evaluation of their electrochemical properties towards clean energy generation†

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To deal with fossil fuel depletion and the rise in global temperatures caused by fossil fuels, cheap and abundant materials are required, in order to fulfill energy demand by developing high-performance fuel cells and electrocatalysts. In this work, a natural organic agent has been used to synthesize nano-structured ZnO/Mn<sub>3</sub>O<sub>4</sub> with high surface area and enhanced electrocatalytic performance. Upon pre-annealing treatment, mixed metal oxide precipitates are formed due to the complex formation between a metal oxide and organic extract. The thermally annealed mixed oxide ZnO/Mn<sub>3</sub>O<sub>4</sub> was characterized by XRD diffractometer, Fourier transform infrared spectroscopy (FTIR), X-ray photoelectron spectroscopy (XPS), field emission scanning electron microscopy (FE-SEM), transmission electron microscopy (TEM) and energy dispersive X-ray spectroscopy (EDX). Gas chromatography-mass spectrometry (GC-MS) identified methyldecylamine as a major stabilizing agent of the synthesized nanomaterial. Using a Tauc plot, the calculated band energy for the synthesized ZnO/Mn<sub>3</sub>O<sub>4</sub> mixed metal oxide was 1.65 eV. Moreover, we have demonstrated the effects of incorporated organic compounds on the surface chemistry, morphology and electrochemical behavior of ZnO/Mn<sub>3</sub>O<sub>4</sub>. The phyto-functionalized ZnO/Mn<sub>3</sub>O<sub>4</sub> was deposited on Ni-foam for electrocatalytic studies. The fabricated electrode revealed good performance with low over-potential and Tafel slope, suggesting it to be suitable as a potential catalyst for water splitting application, in particular for the oxygen evolution reaction (OER). The overall findings of the current study provide a cost-effective and efficient organic template for functionalization and sustainable fabrication of ZnO/Mn<sub>3</sub>O<sub>4</sub> nanomaterial for application as an electrocatalyst.

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## 1. Introduction

Increasing energy demand, the greenhouse effect, and rapid depletion of fossil fuels have motivated researchers towards the development of products designed for the generation of alternative, clean and sustainable energy.<sup>1–113</sup> In order to fulfill energy demands, electrochemical water splitting is one of the eco-friendly and alternative pathways for energy generation.<sup>2,3</sup> The water splitting mechanism is the reverse of the fuel cell reaction and comprises anodic, oxygen evolution reaction (OER) and the cathodic, hydrogen evolution reaction (HER). The

hydrogen evolution reaction on the cathode not only favors a viable route to produce hydrogen from water but is also an important component in the development of fuel cell devices.<sup>4</sup> However, due to sluggish kinetics and a large anodic over-potential in the oxygen evolution reaction (OER), water oxidation is hampered in an alkaline medium.<sup>5–10</sup> To deal with this, significant efforts have been made to design an efficient electrode material which performs best in alkaline medium with a low OER/HER over-potential.<sup>7–12</sup>

Currently, noble metals such as ruthenium oxide (RuO<sub>2</sub>), iridium oxide (IrO<sub>2</sub>), and Pt-based materials are considered as benchmark HER and OER catalysts because of their efficient electro-hindered due to the high cost and low earth abundance. Therefore, it is desirable to explore such bi-functional electrode materials that have low cost, high abundance and high electro-catalytic efficiency with large scale applicability towards OER and HER. Zinc oxide is an interesting transition metal oxide due to its chemical stability, abundance and wide applications in optical and electronic devices making it the best fit for wide application,<sup>13</sup> but its poor electric conductivity

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and high resistance limits its use as electrode material.<sup>14–16</sup> To overcome this problem, a principle approach is to tailor the morphology in order to enhance the surface area of ZnO. It is well known that high surface area improves the charge-transfer capability and enriches the interaction of electrolyte with the electrode surface thus enhancing the diffusion of ions.<sup>17–21</sup> In addition, an alternative to reduce the drawbacks associated with ZnO is to combine it with other metals or metal oxides.<sup>22,23</sup> Doping of different metallic ions and formation of binary mixed metals/metal oxides improves the electric, magnetic, electrochromic and catalytic properties of ZnO.<sup>24–27,106</sup> For the resolution of the challenges associated with ZnO utilization, manganese and manganese-based catalysts are gaining thrust for application in catalytic and fuel cell devices. This is due to their low cost and improved activity compared to Pt-based rare earth metals.<sup>28–32</sup> Therefore, combining Mn oxide with cheap and abundant ZnO is an effective way to improve the electro catalytic activity due to the diversifying the properties of the combination of different components.<sup>33</sup> Previously, binary/mixed metals/metal oxides have been prepared by different physical and the chemical synthesis routes. However, the chemical and physical synthesis method involves chemical reagents or stabilizing agents that are a major concern towards environmental safety and sustainability.<sup>34–36</sup> In this context, successive phyto-synthetic routes have been investigated and reported,<sup>37,38,42–46</sup> and have many advantages in terms of reduced toxicity, and the simplicity of method over the chemical synthesis approach.<sup>39,40,83</sup> In the present study, bio-templated mixed metal oxides have been investigated as a facile, greener nano-material with enhanced surface properties to replace chemically synthesized material. Thus, the focus of the present study is the synthesis of ZnO/Mn<sub>3</sub>O<sub>4</sub> nano-material using leaf extracts of the *Olea ferruginea* Royle (OFR) plant taken from Azad Jammu Kashmir, Pakistan. The selected plant is known for its richness in biologically active compounds *i.e.* phenols, flavonoids and alkaloids.<sup>84,85</sup> Secondly, the plant has not been explored before as a bio reductant in the synthesis of mixed metal oxides. We have tuned the surface morphology of ZnO/Mn<sub>3</sub>O<sub>4</sub> by incorporating carbon and oxygen species as the bioactive compounds aided by the OFR organic extract. Previously, the surface of functionalized nanomaterials has been improved by incorporating carbon, oxygen nitrogen-containing groups using synthetic reagents which enhance the electrochemical performance of catalyst for fuel cell application.<sup>63,86–88</sup> Similarly, related carbon structures including carbon,<sup>89</sup> graphene,<sup>90</sup> carbon fibers<sup>92</sup> and carbon nanotubes<sup>91</sup> have been reported as practicable materials with excellent performance in OER. N-doped carbon materials have been studied extensively for electrochemical studies.<sup>93</sup> The present synthesis approach offers a green synthesis route without the involvement of any auxiliary chemical reagents for the synthesis of ZnO/Mn<sub>3</sub>O<sub>4</sub> nanoparticles. Thus, we believe that the proposed bio template-assisted synthesis route can significantly affect the efficiency of the ZnO/Mn<sub>3</sub>O<sub>4</sub> nano catalyst towards renewable energy to facilitate fast diffusion of ions and current charge carriers.

## 2. Experimental details

*Olea ferruginea* Royle (OFRL) sampling and extract preparation were done by following orthodox protocols.<sup>94–97</sup> Using the organic extract as a reducing and stabilizing agent, ZnO/Mn<sub>3</sub>O<sub>4</sub> nanoparticles (NPs) were synthesized by simply modifying the co-precipitation method (Fig. 1) in which chemical reagents have been used previously.<sup>42–46,98–100</sup> 20 mM salt solution was prepared by mixing zinc(II) acetate dihydrate: Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O and manganese(II) acetate tetrahydrate: Mn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>·4H<sub>2</sub>O separately in 250 ml of distilled water with continuous stirring at room temperature for 20 min. These solutions were combined together with 10 ml of organic extract and the mixture were stirred continuously at 80 °C for an hour. An immediate color change and precipitates formation was observed indicating the initiation of nanoparticles formation. The organic extract aids in the reduction of metal salts to zero-valent metals as presented in Fig. 1. After heating and stirring, the solution was left overnight for incubation at room temperature and then centrifuged at 6000 rpm for 20 minutes. The incorporated organic content results in the stabilization of formed particles leading to the production of dark brown precipitates. Before oven drying at 95 °C, the as-prepared ZnO/Mn<sub>3</sub>O<sub>4</sub> product was washed several times with double-distilled water and then with ethanol. The dried powder was ground to a fine powder and placed in a muffle furnace (D550, Ney Vulcan, USA) at 450 °C to produce ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs.

Phytochemical analysis of the extract and synthesized mixed metal oxide nanoparticles (ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs) were characterized by UV-vis spectroscopy (1602, Biomedical services, Spain) FTIR; Fourier transform infrared spectroscopy (8400, Shimadzu, Japan) and gas chromatography-mass spectroscopy (GC-MS) to identify the organic compounds acting as stabilizing or capping agents. The crystalline phase of the ZnO/Mn<sub>3</sub>O<sub>4</sub> was examined by XRD diffractometer (Bruker Advanced X-ray Solutions D9). X-ray photoelectron spectroscopy (XPS) spectra were collected using a Kratos Axis Ultra Spectrometer (Monochromated Al K $\alpha$  X-ray source,  $h\nu = 1486.6$  eV) to determine surface chemistry. Energy dispersive X-ray spectroscopy (EDX) was used to confirm the purity and the bulk elemental composition of the synthesized material. Structure and morphology were observed by using Quanta-250 FEG scanning electron microscopy (FE-SEM) and FEI Titan G2 80-200 scanning transmission electron microscope (TEM).

Fabrication of a working electrode (ZnO/Mn<sub>3</sub>O<sub>4</sub>) was done by mixing 80 wt% of the synthesized NPs, 10 wt% of acetylene black, and 10 wt% of polyvinylidene difluoride (PVDF) in the presence of *N*-methyl pyrrolidinone (NMP). After homogeneous mixing, the slurry was deposited on Ni-foam and dried at 60 °C under vacuum for 12 hours. Nickel foam (1.5 mm; 1 × 2 cm) was used as the mechanical support to the electro-catalyst (ZnO/Mn<sub>3</sub>O<sub>4</sub>). The mass loading of electrocatalyst was 2 mg cm<sup>-2</sup>.

The catalytic activity of synthesized material was studied using a VersaSTAT 4-500 electrochemical workstation (Princeton Applied Research, USA) in standard electrode configuration with 1 M KOH electrolyte solution. A graphite rod, saturated



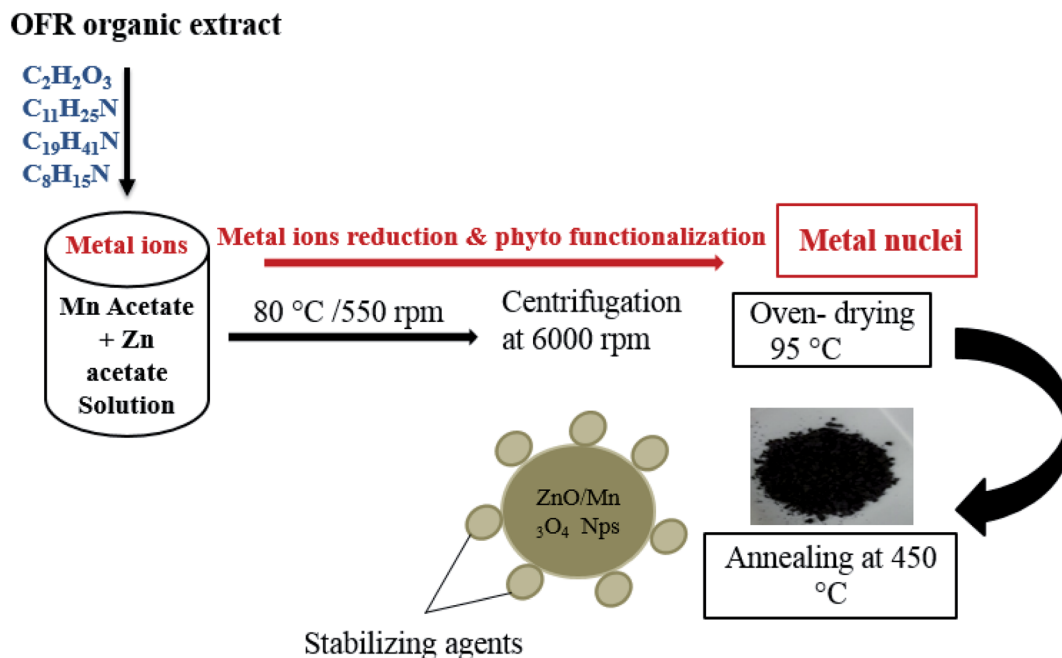


Fig. 1 Schematic diagram of the synthesis procedure and mechanism of phyto-functionalized ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs.

calomel electrode (SCE), and synthesized ZnO/Mn<sub>2</sub>O<sub>3</sub>@Ni-foam were used as a counter, reference and working electrode, respectively. Electro-catalytic testing including linear sweep voltammetry (LSV) and electrochemical impedance spectroscopy (EIS) were performed. LSV was performed at a rate of 1 mV s<sup>-1</sup> for both OER and HER. Using the Nernst equation the potential was converted to RHE (eqn (1)).<sup>41</sup> All EIS measurements were conducted in an applied AC amplitude of 10 mV with the frequency range of 0.05 Hz to 10 kHz.

$$E(\text{RHE}) = E(\text{Ag}/\text{AgCl}) + (0.205 + 0.059 \times \text{pH}) \quad (1)$$

### 3. Result and discussion

The phytochemical analysis and all the relevant detail are given in Fig. S-1, Tables S-1 and S-2.† UV-visible spectroscopy shows absorbance at 271, 327 and 400 nm. The prominent peaks were noticed in the ultra violet and the visible region as shown in the Fig. S-1.† These peaks corresponds to plant phenols, flavonoids and flavonols. The dried powder of the OFR leaf was subjected to FTIR over the range of 400–4000 cm<sup>-1</sup> to determine the chemistry of phyto compounds of the selected plant. Table S-2† summarizes the vibrational frequencies of specific functional groups assigned the aliphatic amines and alkenes present in the leaf powder. The frequency range from 3400–2922 cm<sup>-1</sup> represents the O–H stretching vibration and –C=C– stretch suggesting the presence of alcohols and phenols; alkanes and alkenes, respectively. The frequencies (cm<sup>-1</sup>) 1600 and 1100 suggested the presence of nitro compounds; aromatic amines; alcohols, carboxylic acids, esters and ethers. At frequencies 1030 cm<sup>-1</sup> and 875 cm<sup>-1</sup>, C–N stretch and =C–H bend

attributed to aliphatic amines and alkenes are observed. A weak vibrational mode at 798.56 cm<sup>-1</sup> has been noticed and is assigned to a =C–H bend and C–Cl stretch. GC-MS analysis of the methanolic extract of OFR plant extract identified various compounds as follows. The major bioactive compounds identified in OFR were benzenemethanol, cathinone, 3-azabicyclo nonane, propanedioic acid, 1-cyclohexylethylamine. In agreement with FTIR results, GCMS illustrates majority of the NH-bond containing compounds are amino acids and amines with the highest weight % of cathinone. Cathinone is a mono-amine alkaloid having a ketone group. After cathinone, the highest weight % is of 1-cyclohexylethylamine, benzenemethanol and propanedioic acid. The structures of 1-cyclohexylethylamine, benzenemethanol and propanedioic acid are given in Table S-3† indicate the presence of phenols, flavonoids and amine groups as proposed by spectroscopic analysis.

FTIR of the synthesized nano-powder is carried out to confirm the role of organic content of OFR in the synthesis of NPs. The FTIR spectrum is shown in Fig. 2a that shows vibrational peaks at 2580 cm<sup>-1</sup> (O–H<sub>str</sub>, carboxylic acid), 1104 cm<sup>-1</sup> (CO<sub>str</sub>, alcohols, carboxylic acids, esters, ether) and 958 cm<sup>-1</sup> (C–N<sub>str</sub>, aromatic amines), =C–H bend suggesting alkenes. The inset in Fig. 2a covering 400–600 cm<sup>-1</sup> shows peaks at 412 cm<sup>-1</sup>, 421 cm<sup>-1</sup>, 437 cm<sup>-1</sup>, 491 cm<sup>-1</sup> and 517 cm<sup>-1</sup> corresponding to M–C and M–O (where M = metal ion) bonds suggesting metal oxides of manganese and zinc.

In agreement with the composition of plant extract, GC-MS analysis of ZnO/Mn<sub>3</sub>O<sub>4</sub> confirmed the role of OFR extract as a capping agent in the synthesis. In Fig. 2b the GC-MS chromatogram exhibits a sharp peak at 20.90 retention time corresponding to methyldecylamine (amine derivative) which is an antioxidant present in medicinal plants.<sup>47</sup> It is proposed in the



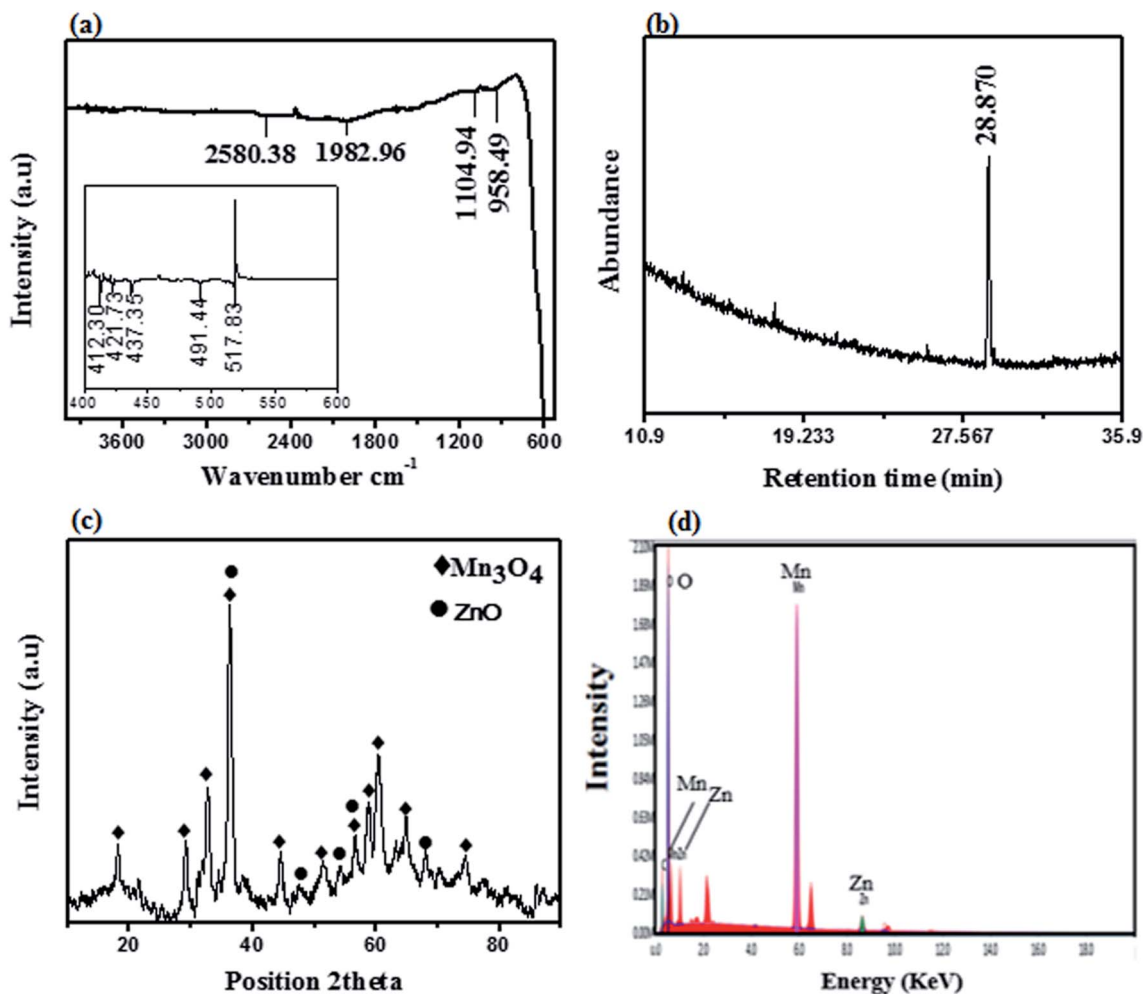


Fig. 2 (a) FTIR spectrum (b) GC-MS chromatogram (c) X-ray diffractogram and (d) EDX spectra of as synthesized ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs.

light of previous studies<sup>63,86,89</sup> that these organic compounds could improve the electrochemical performance of synthesized material by enhancing the electron flow.

The purity and crystallinity of as-synthesized material were examined by powder X-ray diffraction (p-XRD). The XRD pattern of ZnO/Mn<sub>3</sub>O<sub>4</sub> mixed metal oxide is shown in Fig. 2c. The relative peak intensities and position of Mn<sub>3</sub>O<sub>4</sub> and ZnO diffraction peaks are in good agreement with JCPDS card no. 00-024-0734 and 00-036-1451, respectively. Mn<sub>3</sub>O<sub>4</sub> exhibits a tetragonal hausmannite crystal system with cell parameters,  $a = 5.7$ ,  $b = 5.7$ ,  $c = 9.4$  Å. Prominent diffraction peaks indicated by (♦) at  $2\theta$  values of  $18^\circ$ ,  $32.36^\circ$  and  $36.986^\circ$ ,  $44.44^\circ$ ,  $58.5^\circ$  and  $60.634^\circ$  are associated with (101), (103), (211), (220), (321) and (215) planes respectively in good agreement with literature values. ZnO exhibits a hexagonal crystal system with cell parameters of  $a = 3.2$ ,  $b = 3.2$ ,  $c = 5.2$  Å and introduce distinctive characteristic peaks denoted by (\*) at  $2\theta = 36.253^\circ$ ,  $56.60^\circ$  and  $67.96^\circ$  due to reflection of (101), (110) and (112), Miller index planes, respectively.

The average crystallite size was calculated by diffraction line broadening using the Scherrer equation  $D = (0.94\lambda)/(\beta \cos \theta)$  where  $\theta$  is the diffraction angle, the wavelength is represented

by  $\lambda$  ( $\lambda = 1.542$  Å) (Cu-K $\alpha$ ), and  $\beta$  is the FWHM of the diffraction line. The estimated crystallite size for Mn<sub>2</sub>O<sub>3</sub> was 12.43 nm and 11.87 nm for ZnO.

The EDX spectrum is shown in Fig. 2d demonstrating the elemental composition of as-synthesized NPs with atomic % equivalent to the bulk in good agreement with XRD analysis. Through elemental analysis, the purity and uniform composition of nanoparticles have been confirmed. EDX mapping in Fig. 3a–c depicts the uniform distribution of Zn, Mn and O in the whole area. The morphology of synthesized ZnO–Mn<sub>3</sub>O<sub>4</sub> NPs at different magnification has been shown in Fig. 3d–f. The images present polycrystalline structures that could facilitate excellent electrochemical conductivity or catalytic diffusion of ions.<sup>105</sup> Moreover, the voids and spaces between such particle arrangements provide more spaces to foreign ions with enhanced electrochemical performance.<sup>105</sup> The TEM analysis further reveals the morphology of nanostructures at different magnification (Fig. 3g–i).

At 200 nm, nanoclusters have been observed which become clearer in Fig. 3h representing somehow porous morphology which could facilitate the efficient mass transport and electron transfer.<sup>105</sup> In Fig. 3i the nanostructures appear to contain some



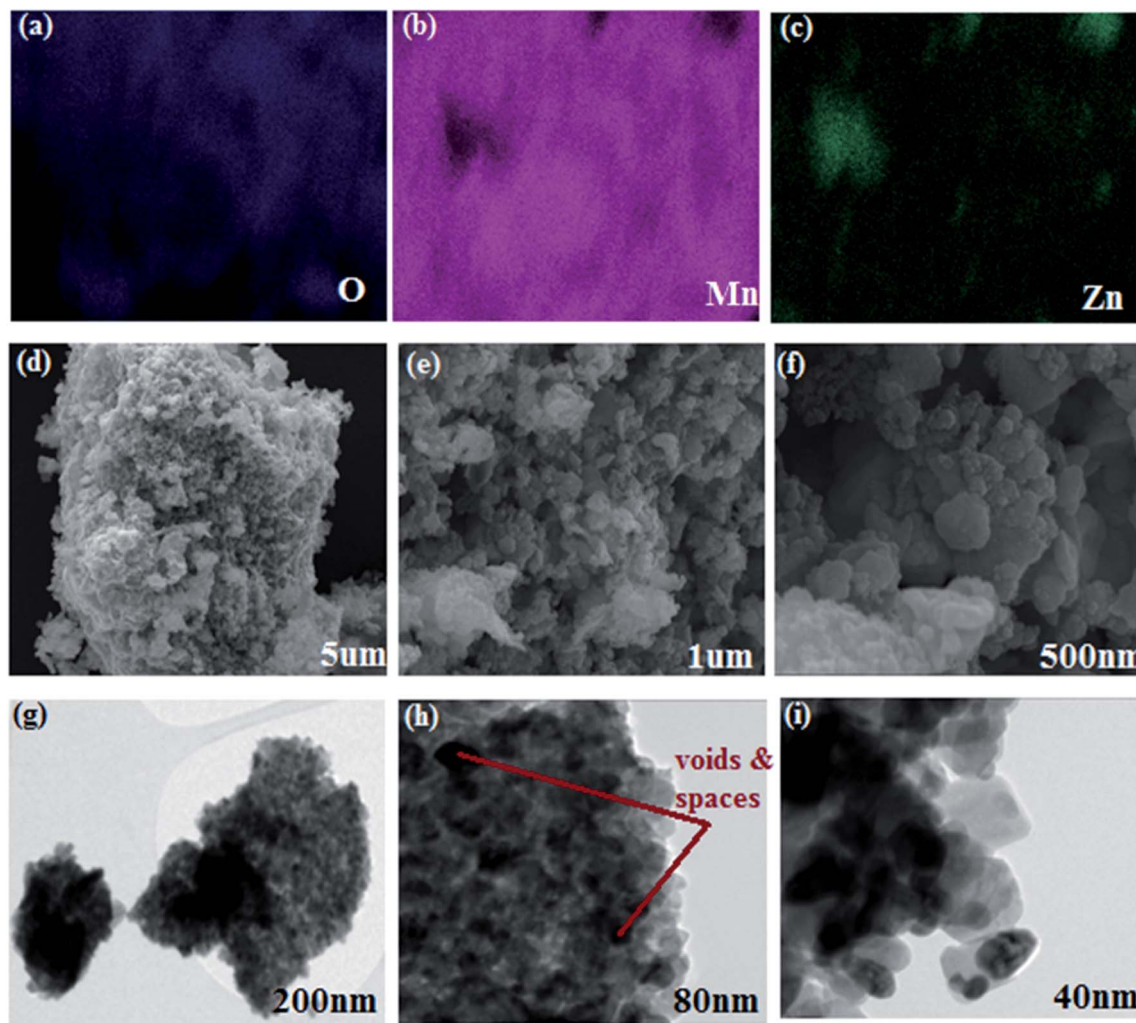


Fig. 3 (a–c) Elemental mapping images of the individual O, Zn, and Mn (d–f) SEM images (g–i) TEM images of ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs at different magnifications.

other nanoparticles inside indicating the formation of mixed structures of both metal oxides ZnO/Mn<sub>3</sub>O<sub>4</sub>. Finally, X-ray photoelectron spectroscopy (XPS) was utilized to examine the surface chemistry and oxidation states of phyto-fabricated ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs (Fig. 4). In the survey spectrum peaks associated with Zn, Mn, O and C can be seen (Fig. 4a). The deconvoluted spectra of Mn 2p spectrum (Fig. 4b) exhibits a doublet with peaks at binding energies of 653.59 eV and 641.78 eV corresponding to the Mn 2p<sub>1/2</sub> and Mn 2p<sub>3/2</sub> as two spin-orbital split peaks<sup>48,101</sup> with splitting energy of 11.81 eV. Fitting of transition metal 2p levels is difficult due to multiplet splitting as the outgoing photoelectron/core hole interacts with the d-electrons in the valence band.<sup>107–109</sup> This becomes even more complex for spinel compounds such as Mn<sub>3</sub>O<sub>4</sub> where there is a mixture of Mn<sup>2+</sup> and Mn<sup>3+</sup> ions in the structure. Biesinger *et al.* have published extensive studies on multiplet splitting of transition metal core levels<sup>109–111</sup> based on the theoretical work of Gupta and Sen.<sup>107,108</sup> However no standard for Mn<sub>3</sub>O<sub>4</sub> is presented in this work, so we use the approach used by Grosvenor for fitting Fe<sub>3</sub>O<sub>4</sub> spectra<sup>111</sup> using a mixture of Mn<sup>2+</sup> and Mn<sup>3+</sup> multiplet

peaks (see ESI Fig. S-2(a and b)†). We find that using the expected ratio of 2 : 1 for Mn<sup>3+</sup> : Mn<sup>2+</sup> gives a reasonable fit compared to a 1 : 1 ratio. Fig. S-2(c) in ESI† also shows the peak fitting for MnO and it is poor, ruling out the possibility that the material is MnO. In Fig. 4b, we show the multiplet fit constrained to the values for Mn<sub>2</sub>O<sub>3</sub> as suggested by Biesinger *et al.*, which gives very good agreement with the experimental data.<sup>109</sup> Although this may appear to be at odds with XRD and EDX data which suggest that the bulk structure is Mn<sub>3</sub>O<sub>4</sub>, we note that XPS is surface sensitive, and at a kinetic energy of ~800 eV we are probing only the top 1 nm. This would be consistent with slight oxidation of the outer surface of the nanoparticles.

Fig. 4c shows the Zn 2p spectra which are fitted well with two peaks corresponding to the spin-orbit split 2p<sub>3/2</sub> and 2p<sub>1/2</sub> peaks at binding energies of 1022.2 eV and 1048.2 eV, respectively, and consistent with the presence of ZnO.<sup>102–104</sup>

The C 1s spectrum shown in Fig. 4d has peaks at binding energies of 284.8 eV (alkyl and aromatic C–C, C–H), 286.3 eV (C–OH, C–O–C and C–N)<sup>52,53</sup> and 288.7 eV (C=O).<sup>112</sup> C 1s peaks can be attributed to the phyto-capping molecules as well as some



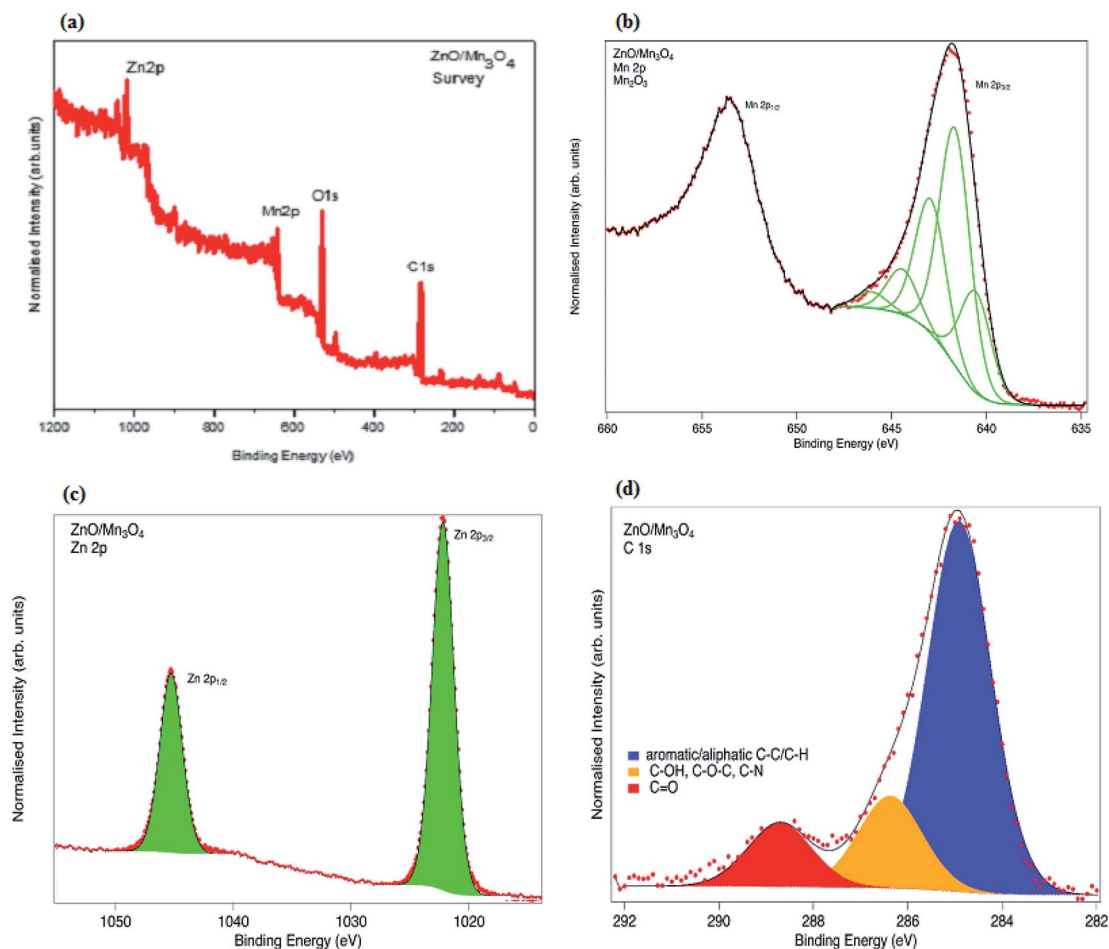


Fig. 4 XPS spectra recorded from ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs: (a) survey spectra, (b) Mn<sub>2</sub>O<sub>3</sub>, (c) Zn 2p, (d) C 1s.

contribution from adventitious carbon. In phyto-directed growth of ZnO/Mn<sub>3</sub>O<sub>4</sub>, residual amounts of organic material are expected because of the role of organic compounds in the synthesis and stabilization of the ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs. In addition it is well established that materials exposed to ambient air are covered by a layer of adventitious carbon, which will also contribute to the observed C 1s signal, particularly the C–C/C–H region at 284.8 eV.<sup>113</sup>

### 3.1. Band gap energy

UV-vis analysis of bio-synthesized NPs has been shown in Fig. 5a. Broad absorption spectra were obtained with absorbance ranging between 300 to 500 nm indicating a red shift. The band energy of the fabricated ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs was calculated from the absorption spectrum by using a Tauc plot as shown in Fig. 5b. The band gap is calculated using the following relation,

$$(\alpha h\nu) = B(h\nu - E_g)^\gamma \quad (2)$$

where  $\alpha$  is the absorption coefficient,  $h\nu$  is the incident photon energy,  $E_g$  is the optical band gap energy,  $B$  is the tailing parameter, and  $\gamma$  is a constant which depends on the band structure of the material. For direct gap materials it is 1/2 and

for indirect gap materials it is 2. The calculated band energy for ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs is 1.65 eV. The band energy value of bulk Mn<sub>3</sub>O<sub>4</sub> (2.70–2.86 eV) decreases with the doping of ZnO which suggests charge transfer properties of the material with enhanced charge carriers.

### 3.2. Electrochemical characterization

The catalytic behavior of the ZnO/Mn<sub>3</sub>O<sub>4</sub> modified Ni-foam catalyst was investigated in 1 M KOH solution at 1 mV s<sup>-1</sup> sweep rate for OER and the results are shown in Fig. 6. Nickel foam (NF) was used as a mechanical support in the present investigation. Linear sweep voltammetry was performed to validate the catalytic performance of synthesized ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs. It was observed that the onset potential (RHE) of oxygen evolution reaction was 1.65 V and the corresponding overpotential was 0.42 V to achieve an OER current density of 10 mA cm<sup>-2</sup> (Fig. 6a). The results have been compared with some reported transition metals based electrocatalysts (Table 1), the obtained over potential in present work is much lower than one of the studies reported by Chung *et al.*, in which mesoporous Mn gave an over potential of 550 mV.<sup>58</sup> At the same time, benchmark catalyst RuO<sub>2</sub> has shown an over potential of 415 mV in a study by Chung *et al.*, which is much lower than the



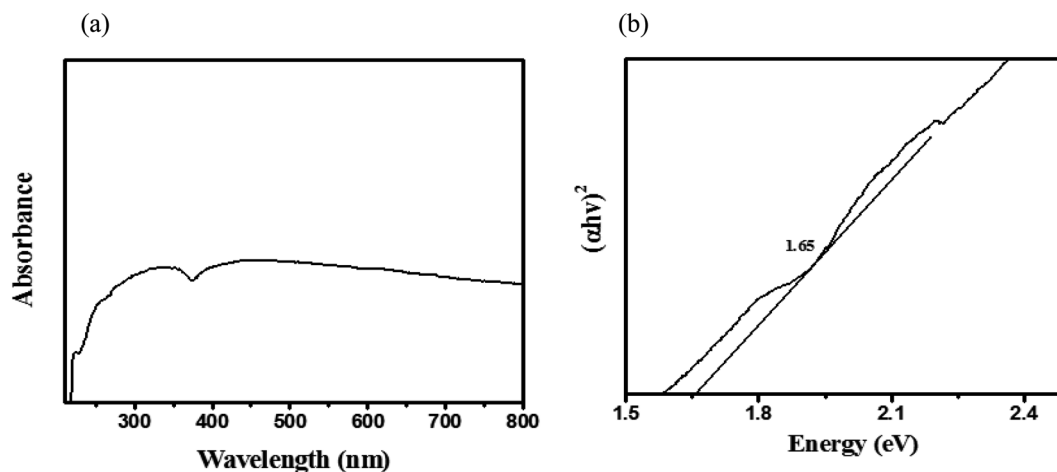


Fig. 5 (a) UV-vis absorption spectra of ZnO/Mn<sub>3</sub>O<sub>4</sub>, (b) band gap of ZnO/Mn<sub>3</sub>O<sub>4</sub> by Tauc plot.

over potential obtained in the present investigation.<sup>58</sup> Commercial RuO<sub>2</sub> and IrO<sub>2</sub> are excellent OER catalyst with the low over-potential of 200 mV (in acid) and 300 mV (in alkaline) to achieve a current density of 10 mA cm<sup>-2</sup> but the only drawback is that the benchmark catalyst required a special setup of proton exchange membrane electrolyzer which is very expensive and rare for large scale application in comparison with the non-

precious transition metal oxide-based mixed metal oxides which are abundant and can easily be operated in alkaline medium.

The uniqueness of the present study is low-cost fabrication of non-precious mixed metal oxide nanoparticles with good electrochemical performance in a completely green and sustainable mode. Another study by Chauhan *et al.*, in which mixed metal

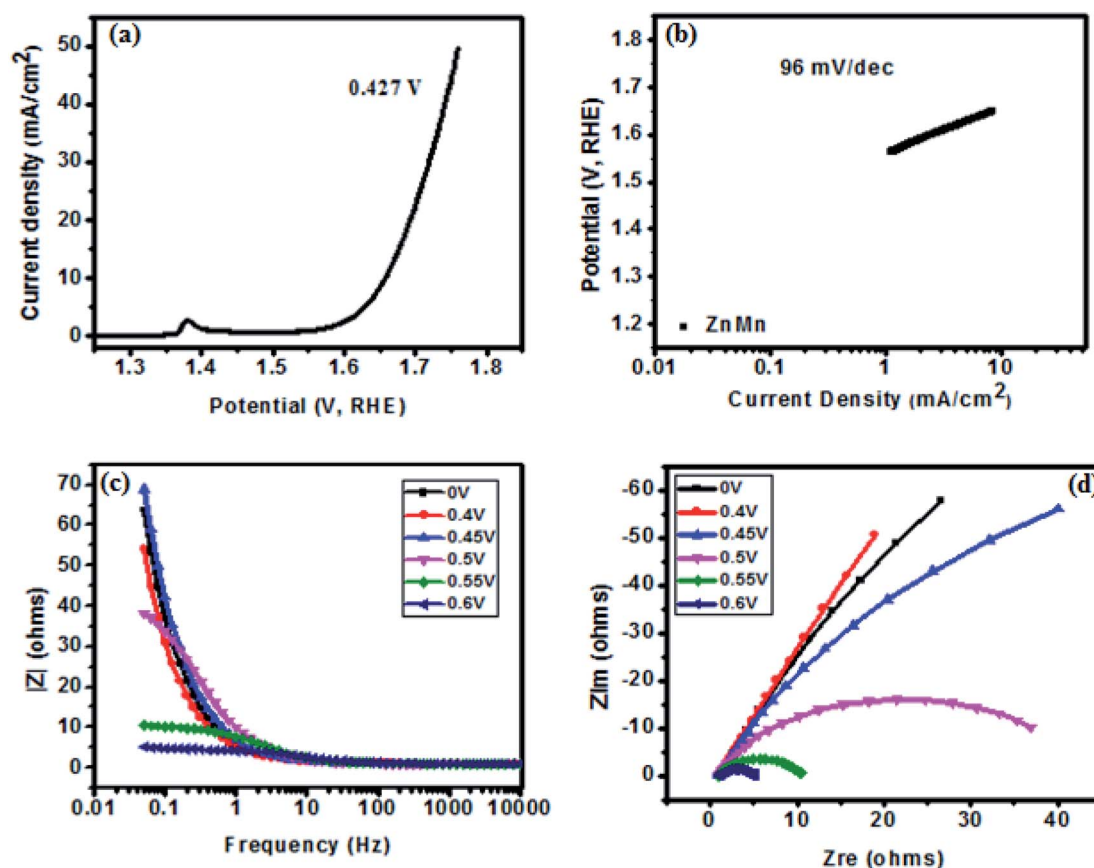


Fig. 6 OER measurements: (a) polarization curves, (b) Tafel slopes, (c) variation of impedance as a function of frequency at various over-potentials (d) Nyquist plot at various over-potentials.



Table 1 Comparison of OER performance of ZnO/Mn<sub>3</sub>O<sub>4</sub> electro catalyst in comparison with transition metal/metal oxides

Catalyst	Over potential	Electrolyte	Tafel slope (mV dec <sup>-1</sup> )	Reference
IrO <sub>2</sub>	320 mV	1 M KOH	—	64
IrO <sub>2</sub>	320 mV	1 M KOH	—	65
CoMnO <sub>2</sub>	390 mV	1 M KOH	95	66
Ni <sub>2</sub> P nanoparticles	500 mV	1 M KOH	—	67
N-doped graphene-NiCo <sub>2</sub> O <sub>4</sub> hybrid paper	434 mV	—	—	68
Co <sub>2</sub> P	370 mV	1 M KOH	128	66
Co <sub>3</sub> O <sub>4</sub> /SWCNTS	550 mV	1 M KOH	104	69
Zn-Co-S nanosheets	390 mV	1 M KOH	136	70
Zn-Co-S nanoplates	330 mV	1 M KOH	97	70
ZnO/Mn <sub>3</sub> O <sub>4</sub> NPs	420 mV	1 M KOH	96	This work

synthesized by chemical route gave very high over potential value as compared to the present study as well as demonstrated a complex synthesis approach.<sup>59</sup> Zhang *et al.* reported a polarization curve of transition metal-based NiO nanoparticles with a very high over potential of 484 mV and 486 mV for NiO@NF-4 and NiO@NF-8 to reach 10 mA cm<sup>-2</sup>.<sup>81</sup> Furthermore, it has been reported by Lv *et al.* in which they demonstrated the synthesis and electrochemical performance of transition metal oxides in comparison with the state of art Pt/C electro catalyst. They proposed the synthesis approach involving the use of polyvinyl pyrrolidone (PVP), sodium thiosulfate (oxidizing agent) and exhibited an over potential of 420 mV, 422 mV and 600 mV for Co<sub>3</sub>O<sub>4</sub> nano cage, NiCo<sub>2</sub>O<sub>4</sub> and 20 wt% Pt/C, respectively.<sup>82</sup> Nevertheless, the present study revealed the environmentally friendly and cost-effective synthesis of ZnO/Mn<sub>3</sub>O<sub>4</sub> low over potential of 0.42 V which outperform the transition metal/metal oxide-based particles synthesized by chemical route involving toxic chemicals and reagents as discussed earlier.

Fig. 6b shows a Tafel plot with current density on the x-axis and potential on the y-axis. The Tafel slope of the fabricated catalyst is a major indicator to demonstrate the electrochemical performance for oxygen evolution. The Tafel slope has been measured from the linear part of the curve potential (RHE) and current density which indicated good OER activity. For studying the kinetics or to reveal the prominent changes in the OER mechanism, the Tafel slope is a very important parameter,<sup>59</sup> since it is inversely correlated to the kinetics of the reaction *i.e.*, smaller Tafel value correlates with faster charge transfer kinetics.<sup>59-61</sup> The value of Tafel slope was 96 mV dec<sup>-1</sup> in this work indicating efficient kinetic behavior of the catalyst for OER. It can also be assessed from the low Tafel value that the synthesized material was well adhered on Ni-foam leading to the observed good OER performance. The Tafel value of fabricated electrode in the current study is lower than the transition non-noble metal-based binary oxides synthesized by Hu *et al.* in which they demonstrated Tafel value of 98 mV dec<sup>-1</sup> and 125 mV dec<sup>-1</sup> for benchmark RuO<sub>2</sub> and binary NiCo@NF respectively.<sup>79</sup> Similarly, Han *et al.* reported standard RuO<sub>2</sub> with a Tafel value of 128 mV dec<sup>-1</sup> (ref. 80) larger than in the present study. Another study by Zheng *et al.* NiOH and IrO<sub>2</sub> showed a Tafel slope of 104 and 118 mV dec<sup>-1</sup>.<sup>81</sup> In comparison with reported studies, the observed Tafel value of 96 mV dec<sup>-1</sup> for

OER in the present investigation is comparatively small which implies that bio-synthesized ZnO/Mn<sub>3</sub>O<sub>4</sub> electro-catalyst exhibit fast reaction kinetics which results in improved OER activity.

Electrochemical impedance spectroscopy (EIS) was performed at varying potential (*V* vs. RHE) to estimate internal resistance (*R<sub>s</sub>*) and charge transfer resistance (*R<sub>ct</sub>*) of the fabricated electrode as shown in Fig. 6d. The cumulative resistance of electrode material, electrolyte and electrolyte/electrode interface comes under *R<sub>s</sub>*.<sup>49,50,54,56</sup> Impedance effect can be subdivided further into high-frequency regions (HFR) and low-frequency regions (LFR). In HFR, the intercept of the semicircle on the real axis denotes *R<sub>s</sub>* and the width of the semicircle graph is representative of *R<sub>ct</sub>*.<sup>49,50,55,56</sup> Whereas in LFR, if the slope of impedance on the real axis is closer to Warburg angle of 45° then that exhibits low Warburg impedance and improved ion transfer or diffusion from the electrolyte to electrode showing high capacitive behavior.<sup>51</sup> It has been observed in present investigation that the width of semicircle is very small at voltage 0.6 V, and estimated *R<sub>s</sub>* and *R<sub>ct</sub>* values were 0.2 Ω and 5 Ω respectively. Similarly the *R<sub>ct</sub>* value at 0.55 V was 10 Ω and increases with decreasing voltage. With decreasing voltage the semicircle becomes larger until the plot become straight at the lowest voltage. It can be assessed that in the HFR the width of the semicircle is very small, indicating the faradaic reaction driven charge transfer resistance. As the Nyquist plot is indicative of charge transfer resistance which represents a transfer of ions/charge across the electrode/electrolyte interface. It means the resistance decreased with increasing potential depicting good electrochemical transfer in terms of acceleration of charge transfer between electrolyte and electrode. The Bode plot (Fig. 6c) between impedance and frequency suggests that impedance shifts towards lower frequency with increasing voltage, exhibiting fast electron conduction of ZnO/Mn<sub>3</sub>O<sub>4</sub> in good agreement with the lower value of *R<sub>ct</sub>*.

One of the prominent aspects of present work is the incorporation of organic functional groups; methyldecylamine (C<sub>11</sub>H<sub>25</sub>N) comprised of C, H, N elements which affect the surface morphology of the synthesized material in term of enhancement of active sites and charge transfer properties of the electro-catalyst as depicted by the XPS analysis in which C-OH, C-O-C compounds have been identified. These results are consistent with previous studies<sup>62,63</sup> which demonstrates that



organic compounds *i.e.* C, H, O and N based groups are mostly derived from bioactive compounds of biomass and organic waste which enhance the catalytic performance of electrode material. Moreover, the nanostructured ZnO/Mn<sub>3</sub>O<sub>4</sub> is suitable for OER as an electro catalyst due to the nano pores as explored in SEM/TEM graphs which give rise to a high surface area for efficient charge transfer in alkaline medium.

The as-synthesized electrode has also been tested for HER studies at varying potential (*V*, RHE), as shown in Fig. 7. The polarization showed an over-potential of 137 mV at  $-20 \text{ mA cm}^{-2}$  at a positive potential of around 0.15. The Tafel slope of ZnO/Mn<sub>3</sub>O<sub>4</sub> is very large for HER as shown in Fig. 7b. The rate of hydrogen evolution is constrained by proton adsorption on active site which cause surface poisoning and lead to deactivation of catalyst. The high value of the Tafel slope exhibits proton ( $\text{H}^+$ ) adsorption on active site (Volmer step) termed as the rate-limiting step.<sup>57,75</sup> As in the present investigation, the HER reaction is sluggish due to the adsorption of protons on the active site. The EIS measurements towards HER have been administered and are shown in Fig. 7d as a Nyquist plot. It was observed that the semicircle increases as the voltage decreases. At a high voltage of 1.4 V, the semicircle arch is small in the high-frequency region and increases with the decrease in voltage, representing a lower resistance and better diffusion efficiency.

In HFR the width of the semicircle is very small indicating the faradaic reaction driven charge transfer resistance. This suggests the good electrochemical performance in term of charge transfer between electrodes as an electrolyte. Some fluctuation in HER result has been observed which might be because of the washing of fabricated electrode or occurrence of some intrinsic reaction due to the alkaline electrolyte. This suggests that HER results is not favorable for this catalyst. OER studies using the as-synthesized electrode material are suitable as an electro catalyst in present work. HER results can be improved by optimizing the conditions. Some of the previous literature on HER has been delineated in Table 2.

It is understood that a different catalyst behaves differently depending upon the choice of electrolyte (acidic or alkaline) as HER catalysis in acidic environment promote more efficient availability of protons as compared to the alkaline medium. The HER mechanism basically comprise three major steps for reaction mechanism which are (i) Volmer step (ii) water dissociation and (iii) formation of active hydrogen intermediates. The dissociation of ions is difficult in alkaline medium as in this case the electro-catalyst required to break the strong H-O-H bond before adsorption of  $\text{H}^+$  on the surface followed by desorption of hydrogen from the surface.<sup>73-75</sup> In present work, in alkaline medium, the material is thought to leach out during dissociation so does not show any HER evolution and resulting in the curve drop down at a positive potential which is ideally

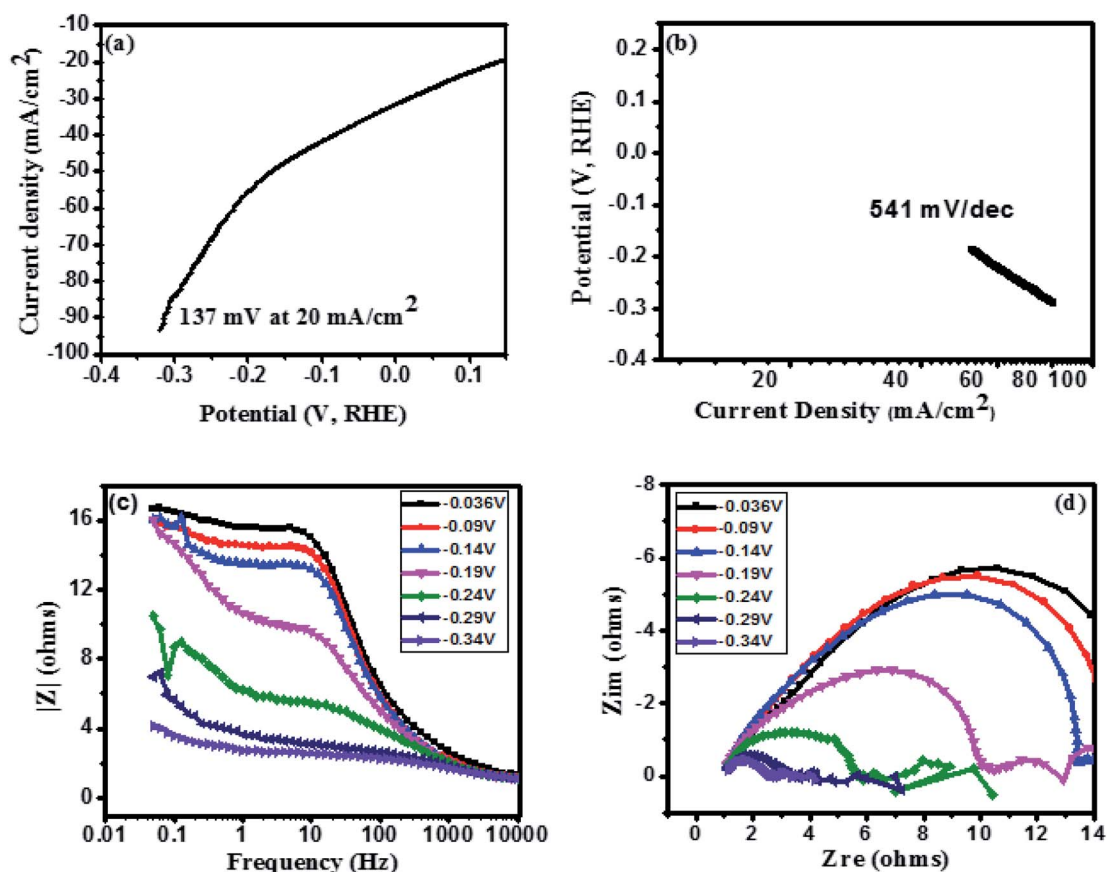


Fig. 7 HER measurements: (a) polarization curves, (b) Tafel slopes, (c) variation of impedance as a function of frequency at various over-potentials (d) Nyquist plot at various over-potentials.



Table 2 HER performance of previously studied transition metal/metal oxides

Catalyst	Over potential (mV)	Tafel slope (mV dec <sup>-1</sup> )	Reference
MnNi	360	—	70
Amorphous MoS <sub>x</sub>	540	—	71
Zn-Co-S nanosheets	413	139	72
NiO@NF	494	317	73

not the property of HER mechanism. The Tafel slope in relation to kinetics of reaction mechanism again explains the mechanism since a high Tafel slope value is obtained which is not ideal for efficient HER. As discussed before it can be concluded that hydrogen adsorption and desorption are the two basic steps for successive H<sub>2</sub> evolution. It means that the catalyst must break the strong bond by forming a strong hydrogen bond with hydrogen atoms and weak bonding with the electrolyte to desorb hydrogen ions from the surface.<sup>76–78</sup> The current work suggests that the ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs in alkaline medium are not efficient for HER, with hydrogen adsorption as the rate-limiting step which either corrodes or over saturates the surface.

## 4. Conclusion

Ni-foam supported ZnO/Mn<sub>3</sub>O<sub>4</sub> was fabricated by phyto-synthesized greener and cost-effective route using the leaf extract of *Olea ferruginea* Royle plant completely devoid of any toxic chemicals utilization for synthetic mode. Prior to electrochemical scrutinization, the as-synthesized material was tested for structural and compositional analysis which confirmed the pure formation of ZnO/Mn<sub>3</sub>O<sub>4</sub> NPs. The electrochemical performance of synthesized material was studied and found that the material could behave as a bifunctional electro-catalyst by optimizing the conditions. But on the basis of current investigation, the material is more suitable for OER performance with low over-potential and low Tafel slope indicating the fast electron transfer. Whereas, the HER results are unsatisfactory. Further, the present method is most favorable in term of commercial and environmental cost as the most greener and environmentally benign electrode material as an alternative to precious metals have been developed. The phyto-functionalized electrode material is a step towards environmental remediation and development which can be set forth in the light of present investigation.

## Conflicts of interest

The authors declare no conflict of interest in this work.

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