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# Foldable water-activated reserve battery with diverse voltages†

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A reserve battery is a device which is inert until its activation and generates electricity by injecting an electrolyte for the purpose of immediate use. Due to a relatively short history and the use in restricted fields, reserve batteries have not attracted attention without any technical advance such as being flexible and foldable. In this study, we demonstrate a way of fabricating a flexible and even foldable reserve battery which is activated by various solutions. A paper electrode composed of cellulose and carbon nanotube, which is working as a cathode, was assembled with a sheet of an aluminum anode. The injection of a NaCl solution resulted in approximately 0.7 V while a KOH solution led to a much higher voltage of 1.3 V than the NaCl electrolyte. Impedance analysis unveiled that the best discharge performance was found in the reserve battery showing the smallest semicircle in impedance spectra, irrespective of electrolytes. And, folding the battery did not degrade the discharge performance, compared with an unfolded battery. Furthermore, the battery cell was even activated by seawater, resulting in about 0.7 V and a nice discharge performance. We think that our battery system can be extended to other reserve batteries requiring flexibility and foldability.

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## 1. Introduction

A battery is a device that produces electricity from chemical reactions and is used in all aspects of daily life. Recently, with growing interest in flexible<sup>1–4</sup> and wearable devices,<sup>5–8</sup> intensive attention has been given to the research and development of batteries as a power source. Generally, a battery can not avoid a self-discharging problem whether it is a primary or secondary battery. Especially, the self-discharge takes place inevitably when batteries are used after long-term storage. A reserve battery can be an alternative to overcome the problem of self-discharge since it is a battery stored in an inactivated state and is activated when the battery needs to be used. The reserve battery can be classified into water-activated,<sup>9–11</sup> gas-activated,<sup>12–14</sup> and heat-activated batteries,<sup>15–17</sup> depending on an activating mechanism. Among them, water-activated batteries are quite attractive because they can be easily activated by injecting fresh water<sup>18</sup> or seawater.<sup>19–23</sup> The water-activated battery was developed in 1940s to satisfy a request for a high-energy-density, long-shelf-life battery for marine applications. It has been known that the reserve battery also has good low-temperature performance, and been used in weather balloons, sonobuoys, radiosondes, electric torpedoes.<sup>24</sup>

With the rapid advances in flexible and wearable devices, much attention was paid to the development of a flexible battery. As a result, many works contributed to the realization of various kinds of flexible batteries.<sup>25–31</sup> However, most successes in a flexible battery are intensively focused on the Li ion battery. Particularly, less interest has been given to a reserve battery, compared with a secondary battery. Furthermore, a flexible and even foldable reserve battery has not been investigated up to now in spite of the merit such as a long shelf life without the self-discharge problem. And, most of the reserve batteries are still restricted to a conventional structure which can not change its shape freely. This situation motivated us to investigate a thin and foldable water-activated reserve battery.

In this work, we demonstrate a way of fabricating a foldable reserve battery activated by injecting a solution as an electrolyte. Among diverse battery systems known as metal–air batteries producing electricity by using metal and oxygen,<sup>32–34</sup> we selected an aluminum–air battery system that usually uses aluminum as an anode and carbon for air cathode. It is because aluminum as well as carbon is cheap, light, and the third abundant element in the earth. Furthermore, it has been known that the aluminum–air battery results in a higher energy density (17 300 W h L<sup>−1</sup>) in comparison with other metal–air batteries (sodium–air: 3 870 W h L<sup>−1</sup>, zinc–air: 6 220 W h L<sup>−1</sup>, lithium–air: 10 600 W h L<sup>−1</sup>).<sup>34</sup> For a battery to be flexible and even foldable, all parts such as an anode, a cathode, and a separator should be bendable without degrading the original properties in their deformed states. To satisfy this, we fabricated a paper electrode composed of multi-walled carbon nanotubes (MWCNTs) and

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cellulose. Previous studies reported that MWCNTs have an electrochemical activity toward oxygen reduction reaction (ORR) in both acidic and alkaline conditions.<sup>35–40</sup> Hence, this electrode was designed to work as a cathode for ORR and to be flexible and foldable freely. Moreover, this paper electrode plays as a role of a reservoir for an electrolyte injected to the reserve battery. As shown in Fig. 1, the single cell of a primary reserve battery was constituted by assembling the paper electrode and a sheet of aluminum with a separator. And, holes were created in the pouch covering the paper electrode for the purpose of both air supply and solution injection. When the solution is injected to the reserve battery, electricity is generated *via* the oxidation of aluminum sheet by the oxygen in the air (anode:  $\text{Al} + 3\text{OH}^- \rightarrow \text{Al}(\text{OH})_3 + 3\text{e}^-$ , cathode:  $\text{O}_2 + 2\text{H}_2\text{O} + 4\text{e}^- \rightarrow 4\text{OH}^-$ ). Depending on the type of a liquid containing different ions, the battery generated different voltages of 0.7 and 1.3 V. Surprisingly, the reserve battery was activated by even seawater, resulting in a voltage of about 0.7 V. Furthermore, folding and crumpling battery did not influence the battery performance.

We think that our primary battery system can be extended to other metal–air batteries for the purpose of flexibility and foldability. Depending on the application, many small cells can be integrated into a single battery by connecting them in series in order to generate a higher voltage than 1.3 V. For example, the battery system suggested in this study can be modulated by varying the size and arrangement of cells, leading to diverse voltage, specific capacity, and power. We expect that when our reserve battery is equipped in a life jacket with a heating element, it can contribute to the delay of hypothermia due to the heat generated by the battery activated by seawater in the sea.

## 2. Experimental

### Fabrication of paper electrode

The paper electrode was prepared by the method described in a previous work.<sup>41</sup> At first, 1000 ml of deionized water was poured into a kitchen mixer grinder. Then, 3 g of shredded waste paper and 0.33 g of MWCNTs were added to the deionized water. The mixture was mechanically ground and mixed for 4 min in the absence of a dispersion agent. Next, 100 ml of the mixture was filtered under vacuum by using a Büchner funnel

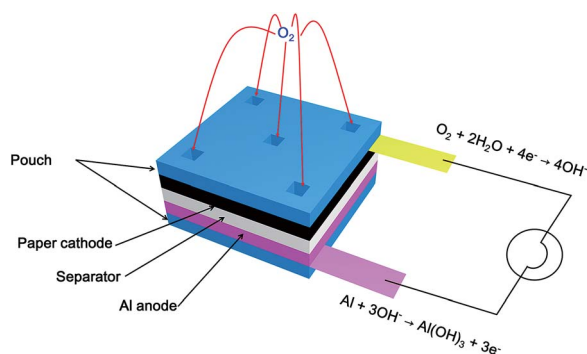


Fig. 1 Schematic illustration of a flexible and foldable reserve battery.

(diameter: 9.5 cm) with filter paper (Advantec 5C, diameter: 9 cm). The filtered mixture was dried and pressed by an office laminator to achieve the paper electrode with an even surface.

### Characterization of paper electrode

Sheet resistance and thickness of the paper electrode were measured by the four-point probe method (Laresa-GP, MCP-T600) and micrometer (Mitutoyo, Tokyo, Japan), respectively. For reproducibility, five sheets of paper were used in the characterization of sheet resistance and thickness. All measurements were performed at five different sites on the paper electrode. Scanning electron microscope (SEM, MIRA 3 FEG SEM) analysis was conducted to observe the morphology of the MWCNTs in the electrode.

### Assembly and activation of reserve batteries

As a first step, aluminum foil (thickness: avg. 17  $\mu\text{m}$ , size: 6 cm  $\times$  6 cm) and the paper electrode (thickness: avg. 110  $\mu\text{m}$ , size: 6 cm  $\times$  6 cm) were assembled with a lab tissue (Kimtech, Yuhan-Kimberly) playing a role as a separator. To collect electrons from the chemical reaction, collectors were placed in both electrodes in the process of cell assembly. Holes were created in the pouch covering the paper electrode for the purpose of air supply as well as electrolyte injection. 1 M and 3 M solutions were prepared by dissolving NaCl and KOH in deionized water to activate the reserve batteries. 400, 600, and 800  $\mu\text{l}$  of the NaCl solution were injected to each battery for activation through the holes. In the case of the KOH solution, 600, 800, and 1000  $\mu\text{l}$  of the electrolyte were used to activate batteries.

### Electrochemical measurement

An electrochemical analyser (ZIVE SP1, WonATech Co., Ltd.) was used to obtain cyclic voltammogram, discharge curve, and impedance spectra. The cyclic voltammetry measurements were conducted in  $\text{N}_2$ - and  $\text{O}_2$ -saturated 0.1 M KOH solution at a scan speed of 20  $\text{mV s}^{-1}$ , respectively. The paper was used as a working electrode. A Pt mesh was adopted as a counter electrode with an Ag/AgCl reference electrode. To evaluate the cell performance, batteries were discharged at a constant current of 3 mA in unfolded and folded states after activation. Impedance was measured in the frequency range from 0.1 and  $1 \times 10^6$  Hz under AC signal amplitude of 10 mV with no applied voltage bias. And, impedance spectra of the full cell were measured after activating each battery in unfolded and folded states.

## 3. Results and discussion

As a first step, we prepared a paper electrode including MWCNTs by the method described in ref. 41. As seen in Fig. 2(a), the color of the paper electrode is black, which means that MWCNTs were evenly composited with the cellulose from waste paper in spite of the absence of a dispersion agent. Moreover, the paper electrode is flexible and even freely foldable as displayed in Fig. 2(a) and (b) without losing the basic properties of paper itself. As explained above, the paper electrode plays a role of a cathode as well as a reservoir to contain the



electrolyte. Thus, keeping the basic properties of paper such as being foldable and wettable has the merit in fabricating a foldable battery. Five sheets of paper were prepared to assure the reproducibility of thickness and sheet resistance. The measurement was conducted at five different sites of each sheet of paper. Fig. 2(c) represents the thickness measured from the five sheets of paper. The black closed square indicates measured data while the red closed circle displays the average value of the measurements. The averages are approximately distributed between 100 and 130  $\mu\text{m}$ , meaning that the preparation method of the electrode is fairly reproducible. The same trend appears in the sheet resistance of the paper electrode as shown in Fig. 2(d). The average values of the sheet resistance of the electrode are ranging between 80 and 100  $\Omega \square^{-1}$ , implying that the paper electrode is quite electrically conductive in spite of the addition of 10 wt% of MWCNTs. In general, the electrical conductivity of a composite material is determined by the degree of percolation when carbon nanotube (CNT) is added as a conductive material. In other words, it can be understood that the superior conductivity of a composite originated from networks where individual MWCNTs are totally connected.

To confirm this, SEM analysis was performed on the surface of the paper electrode. Fig. 3 shows SEM images obtained at low and high magnification enough to show CNTs in the composite. Fig. 3(a) clearly represents the image of a tangled structure of cellulose. This result is in good agreement with previous works reporting the images of the surface of paper or conductive paper to which CNT or carbon fiber was added for electrical conductivity.<sup>41–49</sup> Fig. 3(b) exhibits the SEM image observed from the surface of cellulose in the paper electrode. The image apparently shows that MWCNTs are present on the surface as a structure of tangled networks. The connected MWCNTs are not only a good path for electrons, but also work as a catalyst for ORR. It is obvious that the networked structure contributed to

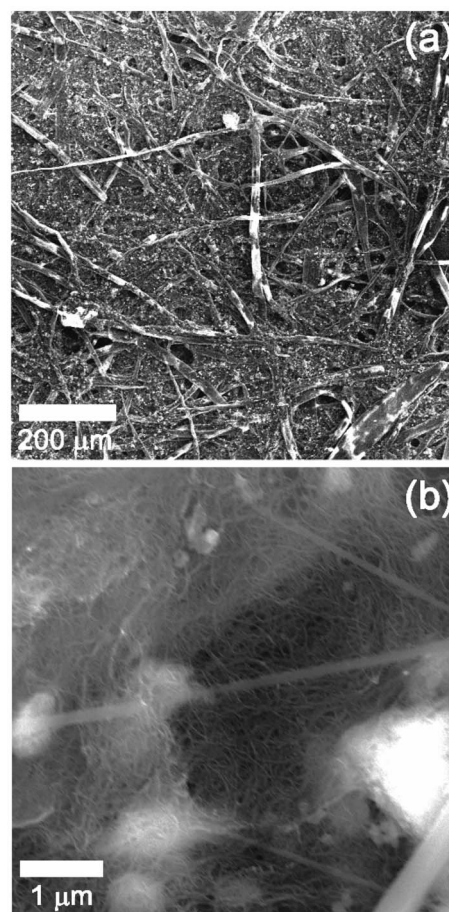


Fig. 3 SEM images of the paper electrode: (a) a tangled structure of cellulose and (b) well-networked MWCNTs on the surface of celluloses. These MWCNTs work as electrical paths for electrons as well as a catalyst for ORR.

the electrical conductivity of the paper electrode, leading to relatively low sheet resistance.

To investigate whether the paper electrode has electrocatalytic activity toward ORR, cyclic voltammetry measurement was carried out in  $\text{N}_2$ - and  $\text{O}_2$ -saturated conditions as shown in Fig. 4. As a result, it has been revealed that ORR occurred around the potential of  $-0.2$  V with respect to an Ag/AgCl electrode. This result is comparable with a previous study<sup>40</sup> investigating the electrocatalytic activity of MWCNTs toward ORR. The study reported that the oxygen reduction onset potential of MWCNTs was  $-0.083$  V with respect to the saturated calomel electrode. Our result indicates that the paper electrode including 10 wt% of MWCNTs can work as a cathode for ORR sufficiently although there exists a little difference in the onset potential, compared with MWCNTs.

Fig. 5 shows photo images of the assembled reserve batteries in normal and deformed shapes. Fig. 5(a) displays a single cell of the reserve battery in an inactivated state. The paper electrode and the aluminum sheet were cut into squares of 6 cm on each side. The assembled battery resulted in a thickness of 335  $\mu\text{m}$  as exhibited in Fig. 5(b). It is because the thickness of all parts is in the range less than 120  $\mu\text{m}$  (paper electrode: avg. 110

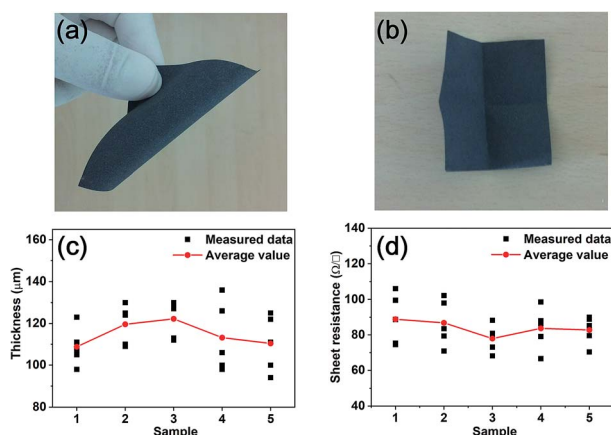


Fig. 2 Photo images of (a) flexible and (b) folded paper electrodes. (c) Thickness and (d) sheet resistance measured from five sheets of paper to confirm the reproducibility of fabrication process. The black closed square and the red closed circle indicate measured data at five different sites of the paper electrode and the average values of the acquired data, respectively.



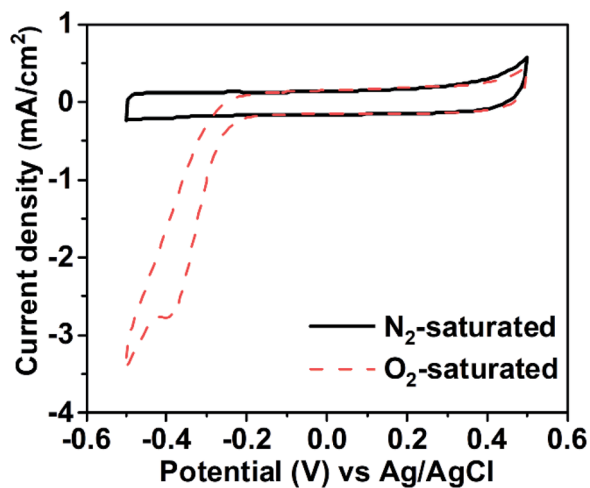


Fig. 4 Cyclic voltammetry curves of the paper cathode in (black solid line) N<sub>2</sub>- and (red dashed line) O<sub>2</sub>-saturated 0.1 M KOH solution.

$\mu\text{m}$ , aluminum sheet: avg. 17  $\mu\text{m}$ , separator: avg. 76  $\mu\text{m}$ ). This thickness can be controlled by changing each part to thinner or thicker one. Fig. 5(c) presents the reserve battery activated by injecting seawater. The output voltage of 0.7 V was measured immediately as soon as seawater permeated the paper electrode and the separator after injection. The discharge performance of the reserve battery activated by seawater will be discussed later. When the battery was activated by the KOH solution, the voltage increased up to 1.3 V as shown in Fig. 5(d), which is enough to trigger a desk clock (see the ESI Movie 1†). These results clarify that different voltages are generated by injecting a dissimilar solution. To investigate the deforming effect on the voltage, the battery activated by the KOH solution was folded and clipped as seen in Fig. 5(e). Interestingly, we can see that the voltage is not

drastically changed, compared with the battery activated by the same solution in an unfolded state. Then, the battery was severely crumpled to see how the voltage changes under harsh conditions. Fig. 5(f) represents that the battery can be crumpled successfully like paper, and the voltage generated from the cell was measured to be 1.2 V which is a close value to the ones obtained from unfolded and folded batteries. The severely crumpled reserve battery also triggered the desk clock (see the ESI Movie 2†). It seems that crumpling battery does not influence the performance of the reserve battery significantly. Furthermore, the battery triggered the desk clock underwater for 1.75 h despite the fact that the system is fully open to water (see the ESI Movie 3†). Fig. 5(g) shows an exemplary application of our battery system where four cells are connected in series and packed into a single cell *via* a lamination pouch by using a transparent laminator. In this case, the paper electrode and the aluminum were cut into squares of 3 cm. The battery was activated by injecting the KOH solution through holes created in the transparent pouch and generated a voltage of 5.2 V. Considering that a single cell activated by the KOH solution generated a voltage of 1.3 V, this is a quite reasonable output voltage.

After preparing an empty cell, we should investigate how much an electrolyte is necessary for maximum discharge performance. For this, we observed a change in impedance spectra obtained from the reserve battery activated by different volumes of each solution. Fig. 6(a) and (b) show the impedance spectra measured from the batteries activated by the 1 M and 3 M NaCl solutions, respectively. In both cases, the volumes of 400, 600, and 800  $\mu\text{l}$  were injected to an empty cell as an electrolyte. When 400  $\mu\text{l}$  of the 1 M NaCl solution was used for activation, the largest semicircle was measured in the spectra. Then, the injection of 600  $\mu\text{l}$  resulted in a relatively small semicircle, compared with 400  $\mu\text{l}$ . When the volume of the

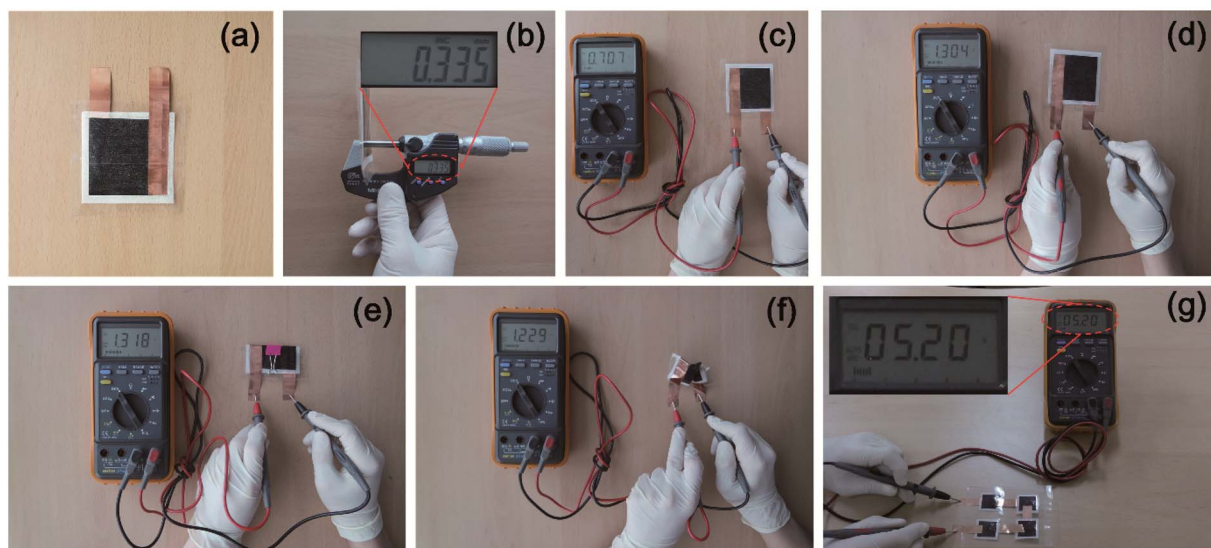


Fig. 5 (a) An assembled single cell of the reserve battery (size: 6 cm  $\times$  6 cm). (b) The thickness of the single cell. Different voltages generated by electrolytes of both (c) seawater and (d) the KOH solution. The battery can be (e) folded and (f) even severely crumpled without degrading the voltage greatly. (g) Integrated four cells in one pouch to generate a high voltage of 5.2 V. The size of each cell is 3 cm  $\times$  3 cm.



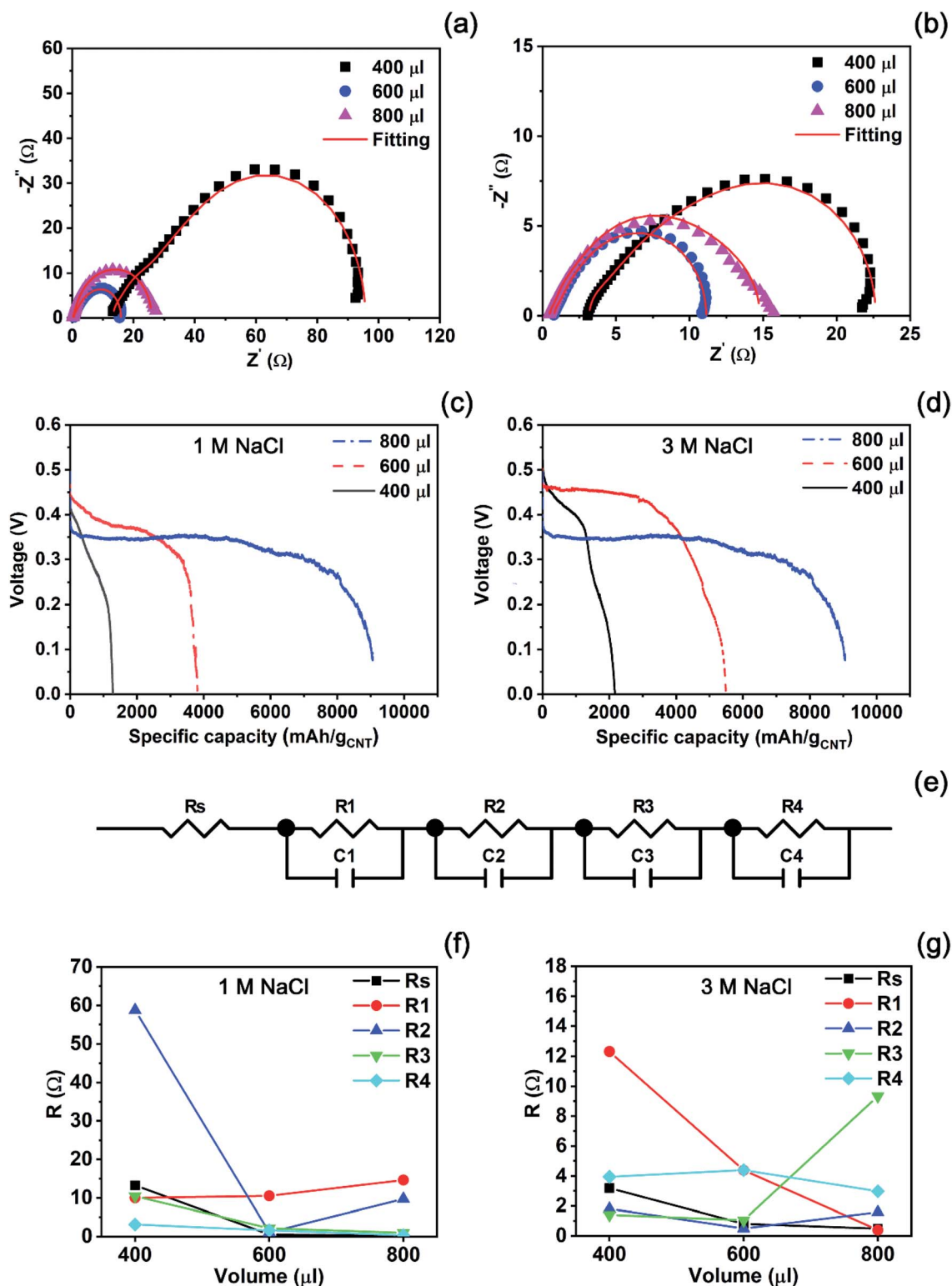


Fig. 6 (a and b) Impedance spectra and (c and d) discharge curves of batteries activated by the 1 M and 3 M NaCl solutions at different volumes. (e) ECM fitting all experimental data nicely shown in (a) and (b). A change in fitted parameters of the ECM for the impedance of the batteries started by (f) the 1 M and (g) 3 M NaCl solutions.

electrolyte increased up to 800  $\mu\text{l}$ , the semicircle became larger than the one recorded from the battery started by 600  $\mu\text{l}$  of the 1 M NaCl solution. The diameter of the semicircle corresponds

to a polarization resistance. Hence, a small semicircle implies that internal resistance becomes small and results in high cell performance. From this, we can expect that the injection of 600



$\mu\text{l}$  will result in the highest performance among the experimental conditions. In the case of the 3 M NaCl solution, the same trend appeared as seen in Fig. 6(b). The smallest semicircle exhibited at the volume of 600  $\mu\text{l}$  of the electrolyte while the largest one was recorded at the volume of 400  $\mu\text{l}$ . The semicircle measured from the reserve battery activated by 800  $\mu\text{l}$  was located between the largest and the smallest one. Therefore, the best performance is expected at the volume of 600  $\mu\text{l}$  within the experimental range.

In order to confirm if the highest battery performance is obtained at the lowest impedance, we conducted a discharge test at a constant current of 3 mA after activating empty batteries by each volume of the electrolytes as presented in Fig. 6(c) and (d). When the 1 M NaCl solution activated the batteries, the specific capacity of each battery increased in proportion to the volume of the solution as plotted in Fig. 6(c). The 400  $\mu\text{l}$  of the electrolyte resulted in 1278 mA h  $g_{\text{CNT}}^{-1}$ , and the specific capacity increased to 3806 mA h  $g_{\text{CNT}}^{-1}$  at the injection volume of 600  $\mu\text{l}$ . The highest specific capacity of 9057 mA h  $g_{\text{CNT}}^{-1}$  was obtained at the volume of 800  $\mu\text{l}$ . However, we can see a relatively low voltage during discharge in Fig. 6(c), compared with other two cases. The figure presents that the best performance in the aspect of output voltage exhibits when the battery is activated by 600  $\mu\text{l}$  of the 1 M NaCl solution. In the case that the 3 M NaCl solution was used to activate batteries, this phenomenon becomes clear as seen in Fig. 6(d). We can see that the specific capacity increased from 2165 to 5489 mA h  $g_{\text{CNT}}^{-1}$  as the injection volume changed from 400 to 600  $\mu\text{l}$ . The injection volume of 800  $\mu\text{l}$  to the reserve battery led to the specific capacity of 9057 mA h  $g_{\text{CNT}}^{-1}$  which is greater than the value from the battery by 600  $\mu\text{l}$  of the electrolyte. However, the output voltage during discharging was much lower than other voltages.

To examine why battery performance is influenced by the volumes of electrolytes, we found an equivalent circle model (ECM) fitting all impedance data plotted in Fig. 6(a) and (b). Fig. 6(e) presents the unique ECM comprised of both resistance and capacitance elements with solution resistance ( $R_s$ ). All fitted parameter values were summarized in Table S1 (see the ESI†). In the case of the 1 M NaCl solution, most of resistance elements except  $R_1$  decrease as shown in Fig. 6(f) when the volume increases from 400 to 600  $\mu\text{l}$ . As the volume of the electrolyte exceeds 600  $\mu\text{l}$ ,  $R_1$  and  $R_2$  increase at the volume of 800  $\mu\text{l}$ . Therefore, it can be understood that the injection of 800  $\mu\text{l}$  of the 1 M NaCl solution increased the internal resistance, resulting in a low output voltage. Fig. 6(g) reveals that the same trend is seen at high concentration of the 3 M NaCl solution. Most of resistance elements such as  $R_s$ ,  $R_1$ ,  $R_2$ , and  $R_3$  decrease when the injection volume increases from 400 to 600  $\mu\text{l}$ . Then, the resistance elements of  $R_3$  increase abruptly at the volume of 800  $\mu\text{l}$ . From these results, it can be interpreted that although the excess volume of the electrolyte results in high specific capacity, the increase in resistance values leads to low voltage during discharge irrespective of concentration.

Fig. 5 showed that the KOH solution generated a higher voltage than the NaCl solution. Based on this preliminary result, we prepared each of 1 M and 3 M KOH solutions to see how the

discharge performance of the battery varies with concentration at a high voltage. Before a discharge test, we also conducted impedance analysis to determine the optimal amount of the KOH solution which leads to the lowest impedance in the experimental conditions. Fig. 7(a) and (b) show impedance spectra measured from the batteries activated by each of 1 M and 3 M KOH solutions, respectively. In the case of the 1 M solution, a semicircle becomes small gradually as the injection volume increases from 600 to 1000  $\mu\text{l}$ . Finally, the smallest semicircle appears at the volume of 1000  $\mu\text{l}$  as seen in Fig. 7(a). Also, Fig. 7(b) presents that the batteries activated by 600, 800, and 1000  $\mu\text{l}$  of the 3 M KOH solution exhibit the smallest semicircle at the volume of 1000  $\mu\text{l}$ , which is similar with the case of the 1 M KOH solution.

Also, a discharge test of the batteries activated by the KOH solution was carried out at a constant current of 3 mA as seen in Fig. 7(c) and (d). We can see that a high voltage is generated in the discharge test, compared with the activated ones by NaCl solutions. When the battery was activated by the 1 M KOH solution, the specific capacity of the battery increased in proportion to the injected volume. 600  $\mu\text{l}$  of the 1 M KOH solution resulted in the specific capacity of 275 mA h  $g_{\text{CNT}}^{-1}$ . Then, it finally approached 874 mA h  $g_{\text{CNT}}^{-1}$  at the volume of 1000  $\mu\text{l}$ , which is consistent with the impedance analysis. That is, the smallest semicircle ensures the highest discharging performance originated from the optimal volume of the electrolyte. When the concentration increased to 3 M, the injection of 600 and 800  $\mu\text{l}$  to reserve batteries resulted in the specific capacity values of 640 and 1421 mA h  $g_{\text{CNT}}^{-1}$  as shown in Fig. 7(d). Then, the highest specific capacity of 1867 mA h  $g_{\text{CNT}}^{-1}$  was achieved at the volume of 1000  $\mu\text{l}$  which is the end point in the experimental range.

For further analysis, we investigated the unique ECM as seen in Fig. 7(e) which fits all impedance data plotted in Fig. 7(a) and (b). Fitted parameters were summarized in Table S2.† When the battery is activated by the 1 M KOH solution, most of resistance elements except  $R_2$  decrease at the volume of 1000  $\mu\text{l}$  as plotted in Fig. 7(f). However, the  $R_2$  element did not show a rapid change at each volume, implying that the increase of an internal resistance by  $R_2$  does not influence the discharge performance at the volume of 1000  $\mu\text{l}$ . As the concentration of the electrolyte rises to 3 M, the highest discharge performance exhibited at the volume of 1000  $\mu\text{l}$  in the experimental range. In Fig. 7(g), resistance elements except  $R_1$  decrease when the injection volume increases from 600 to 1000  $\mu\text{l}$ . However, the variation in the element of  $R_1$  is not remarkable in the experimental range, compared with other resistance elements. Therefore, it can be interpreted that in the case of KOH solution, the increase of resistance element is not influential to the discharge performance of a reserve battery at the volume of 1000  $\mu\text{l}$ .

As shown in Fig. 5(c), the reserve battery was easily activated by seawater, resulting in approximately 0.7 V. To examine whether seawater can generate electricity as much as other electrolytes do, we carried out a discharge test under the same condition. 600  $\mu\text{l}$  of seawater shipped from East Sea was injected into an inactivated reserve battery without any treatment. As



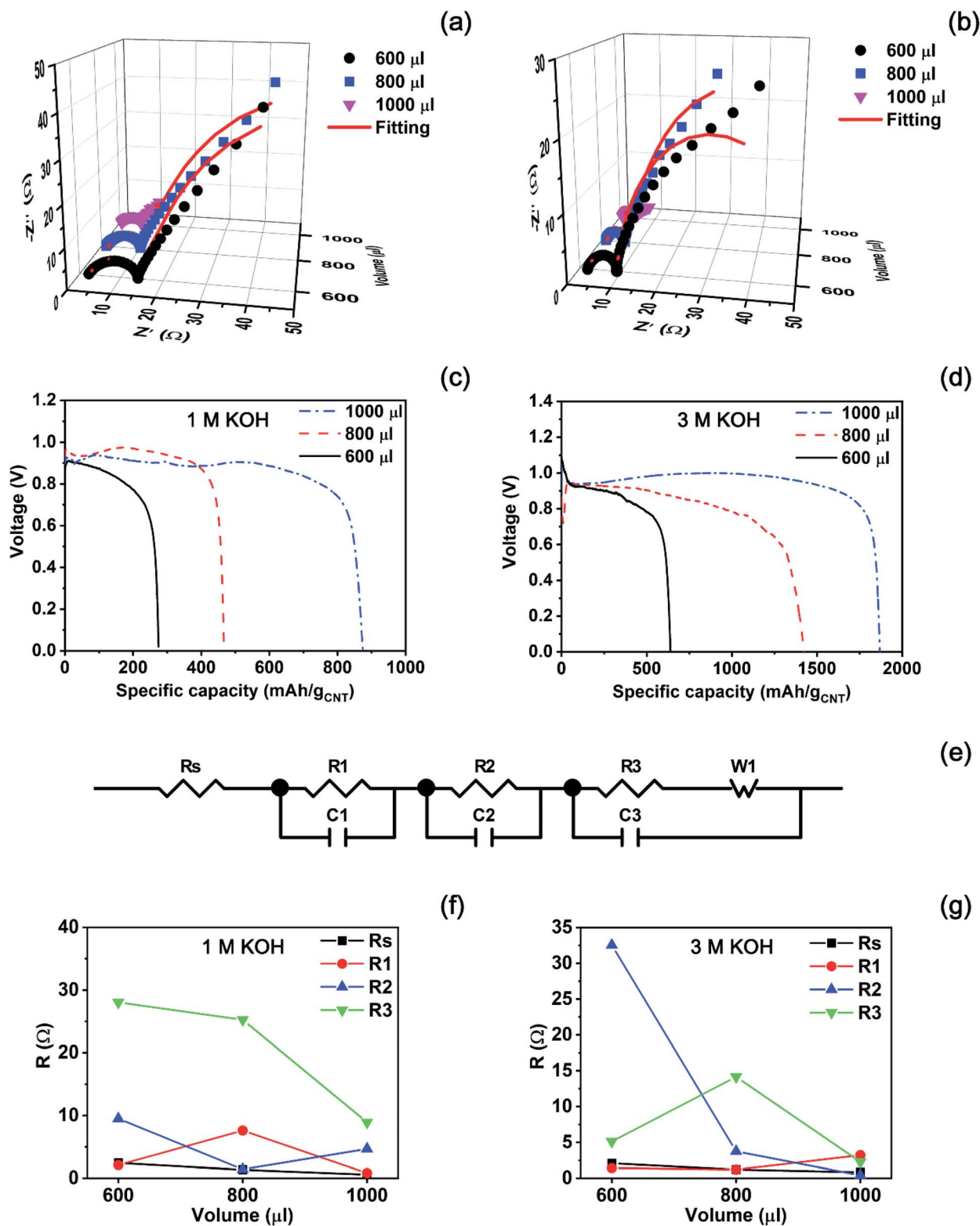


Fig. 7 (a and b) Impedance spectra and (c and d) discharge curves of batteries activated by the 1 M and 3 M KOH solutions at different volumes. (e) ECM fitting all experimental data nicely shown in (a) and (b). A change in fitted parameters of the ECM for the impedance of the batteries started by (f) the 1 M and (g) 3 M KOH solutions.

a result, a nice discharging curve was obtained and specific capacity approached  $8389 \text{ mA h g}_{\text{CNT}}^{-1}$  as seen in Fig. S1 (see the ESI†). This value is quite comparable with the results of

reserve batteries activated by the 1 M and 3 M NaCl solutions, implying that empty cells of reserve batteries can be used in the ocean for the purpose of emergency rescue.



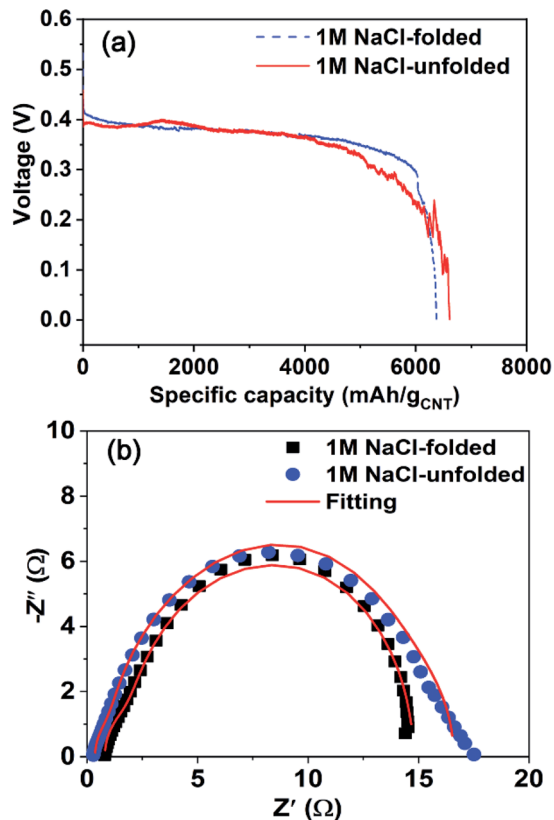


Fig. 8 (a) Discharge performance at a constant rate of 3 mA and (b) impedance spectra of unfolded and folded batteries activated by the 1 M NaCl solution.

In order to study if folding reserve batteries influences the discharge performance, the batteries activated by 700  $\mu\text{l}$  of 1 M NaCl were discharged at a constant current of 3 mA as seen in Fig. 8(a). It seems that the folding effect did not degrade the battery performance significantly, compared with an unfolded battery activated and discharged under the same experimental conditions. We can see that the folded battery rather performs better than the unfolded one in the aspect of the voltage during discharge. The reason can be also understood by examining the impedance spectra measured in folded and unfolded states. Fig. 8(b) presents that the battery in a folded state exhibits a relatively small semicircle in impedance analysis, meaning that the effect of folding a battery is not influential to discharge performance. The parameters estimated by using the ECM in Fig. 6(e) were summarized in Table S3 (see the ESI†). These results strongly show that it is possible to fabricate a battery which is freely transformed without any degradation in the battery performance.

## 4. Conclusions

In summary, we demonstrated a way of fabricating the extremely thin and foldable reserve battery which can be activated by not only the KOH and NaCl solutions, but also seawater. The paper electrode composed of MWCNTs and cellulose played a role of a cathode for ORR and a reservoir to

preserve the injected electrolytes, enabling the battery to be folded and crumpled severely. The empty cell generated different voltages, depending on the type of the injected solution. The injection of the KOH solution resulted in a relatively high voltage of 1.3 V while the battery initiated by the NaCl solution led to 0.7 V. Impedance analysis pointed out that the internal resistance of the reserve battery were influenced by the amount of the injected solution. Consequently, the best discharge performance was achieved at the volume of electrolytes leading to the lowest impedance. Unfolded and folded batteries exhibited almost same semicircle in impedance analysis, supporting that folding battery did not degrade the discharge performance.

The novel battery system suggested here can be extended to other metal–air batteries for the purpose of both flexibility and foldability.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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## Notes and references

- 1 T.-H. Han, Y. Lee, M.-R. Choi, S.-H. Woo, S.-H. Bae, B. H. Hong, J.-H. Ahn and T.-W. Lee, *Nat. Photonics*, 2012, **6**, 105–110.
- 2 S. J. Park, O. S. Kwon, S. H. Lee, H. S. Song, T. H. Park and J. Jang, *Nano Lett.*, 2012, **12**, 5082–5090.
- 3 B. Tian, T. Cohen-Karni, Q. Qing, X. Duan, P. Xie and C. M. Lieber, *Science*, 2010, **329**, 830–834.
- 4 S. Kim, H.-J. Kwon, S. Lee, H. Shim, Y. Chun, W. Choi, J. Kwack, D. Han, M. Song, S. Kim, S. Mohammadi, I. Kee and S. Y. Lee, *Adv. Mater.*, 2011, **23**, 3511–3516.
- 5 Y.-H. Lee, J.-S. Kim, J. Noh, I. Lee, H. J. Kim, S. Choi, J. Seo, S. Jeon, T.-S. Kim, J.-Y. Lee and J. W. Choi, *Nano Lett.*, 2013, **13**, 5753–5761.
- 6 J. Ren, Y. Zhang, W. Bai, X. Chen, Z. Zhang, X. Fang, W. Weng, Y. Wang and H. Peng, *Angew. Chem., Int. Ed.*, 2014, **53**, 7864–7869.
- 7 J. J. Park, W. J. Hyun, S. C. Mun, Y. T. Park and O. O. Park, *ACS Appl. Mater. Interfaces*, 2015, **7**, 6317–6324.
- 8 Y. Wang, L. Wang, T. Yang, X. Li, X. Zang, M. Zhu, K. Wang, D. Wu and H. Zhu, *Adv. Funct. Mater.*, 2014, **24**, 4666–4670.
- 9 W. N. Carson, N. Y. Schenectady and W. H. Fischer, *Sea water-activated primary battery*, *US Pat.* 3352718, Nov. 14, 1967.



- 10 J. B. Mullen and P. L. Howard, *J. Electrochem. Soc.*, 1946, **90**, 529–544.
- 11 F. Sammoura, K. B. Lee and L. Lin, *Sens. Actuators, A*, 2004, **111**, 79–86.
- 12 W. G. Darland and M. T. Ross, Gas-depolarized battery, *US Pat.* 2914595, Nov. 24, 1959.
- 13 M. R. Hatfield, Heavy duty gas depolarized dry battery, *US Pat.* 2615931, Oct. 28, 1952.
- 14 R. E. Johnson, W. G. Darland, C. A. Grulke, N. C. Cahoon and H. F. Schaefer, Gas-activated primary cells, *US Pat.* 2948767, Aug. 9, 1960.
- 15 Y. Hu, H. Li, X. Huang and L. Chen, *Electrochem. Commun.*, 2004, **6**, 28–32.
- 16 R. T. Carlin, H. C. D. Long, J. Fuller and P. C. Trulove, *J. Electrochem. Soc.*, 1994, **141**, L73–L76.
- 17 B. Garcia, S. Lavalley, G. Perron, C. Michot and M. Armand, *Electrochim. Acta*, 2004, **49**, 4583–4588.
- 18 K. Vuorilehto, *J. Appl. Electrochem.*, 2003, **33**, 15–21.
- 19 L. Wen, K. Yu, H. Xiong, Y. Dai, S. Yang, X. Qiao, F. Teng and S. Fan, *Electrochim. Acta*, 2016, **194**, 40–51.
- 20 J. Zhao, K. Yu, Y. Hu, S. Li, X. Tan, F. Chen and Z. Yu, *Electrochim. Acta*, 2011, **56**, 8224–8231.
- 21 R. Renuka, *Mater. Chem. Phys.*, 1999, **59**, 42–48.
- 22 Y. Shi, C. Peng, Y. Feng, R. Wang and N. Wang, *Mater. Des.*, 2017, **124**, 24–33.
- 23 K. Yu, X. Tan, Y. Hu, F. Chen and S. Li, *Corros. Sci.*, 2011, **53**, 2035–2040.
- 24 D. Linden and T. B. Reddy, Reserve batteries, *Handbook of Batteries*, McGraw-Hill, 2001, 3rd edn, p. 16.3.
- 25 H. Li, C. Han, Y. Huang, Y. Huang, M. Zhu, Z. Pei, Q. Xue, Z. Wang, Z. Liu, Z. Tang, Y. Wang, F. Kang, B. Li and C. Zhi, *Energy Environ. Sci.*, 2018, **11**, 941–951.
- 26 M. Koo, K.-I. Park, S. H. Lee, M. Suh, D. Y. Jeon, J. W. Choi, K. Kang and K. J. Lee, *Nano Lett.*, 2012, **12**, 4810–4816.
- 27 N. Li, Z. Chen, W. Ren, F. Li and H.-M. Cheng, *Proc. Natl. Acad. Sci.*, 2012, **109**, 17360–17365.
- 28 Y. Huang, J. Liu, J. Zhang, S. Jin, Y. Jiang, S. Zhang, Z. Li, C. Zhi, G. Due and H. Zhou, *RSC Adv.*, 2019, **9**, 16313–16319.
- 29 M. Kammoun, S. Berg and H. Ardebili, *Nanoscale*, 2015, **7**, 17516–17522.
- 30 Q. Li and H. Ardebili, *J. Power Sources*, 2016, **303**, 17–21.
- 31 Y. Li, D. Ye, B. Shi, W. Liu, R. Guo, H. Pei and J. Xie, *Phys. Chem. Chem. Phys.*, 2017, **19**, 7498–7505.
- 32 J. Ma, C. Qin, Y. Li, F. Ren, Y. Liu and G. Wang, *J. Power Sources*, 2019, **430**, 244–251.
- 33 Y.-J. Wang, B. Fang, D. Zhang, A. Li, D. P. Wilkinson, A. Ignaszak, L. Zhang and J. Zhang, *Electrochem. Energy Rev.*, 2018, **1**, 1–34.
- 34 Md. A. Rahman, X. Wang and C. Wen, *J. Electrochem. Soc.*, 2013, **160**, A1759–A1771.
- 35 Y. Li, W. Xia, R. Zou, J. Zhang, Z. Chen and Q. Xu, *RSC Adv.*, 2015, **5**, 96580–96586.
- 36 X. Wu, Z. Liu, Y. Jiang, J. Zeng and S. Liao, *J. Mater. Sci.*, 2017, **52**, 8432–8443.
- 37 H.-J. Zhang, H. Li, X. Li, B. Zhao and J. Yang, *Int. J. Hydrogen Energy*, 2014, **39**, 16964–16975.
- 38 L. Osmieri, A. H. A. M. Videla and S. Specchia, *J. Power Sources*, 2015, **278**, 296–307.
- 39 Z. Wang, H. Lei, R. Cao and M. Zhang, *Electrochim. Acta*, 2015, **171**, 81–88.
- 40 H.-J. Zhang, H. Li, C. Deng, B. Zhao and J. Yang, *ECS Electrochem. Lett.*, 2015, **4**, H33–H37.
- 41 D.-H. Kim, Y.-I. Cho, J. H. Choi, H.-S. Kim, H. C. Shin, T. S. Lee, J. W. Jung, H.-D. Kim, D.-J. Lee and G. T. Kim, *RSC Adv.*, 2015, **5**, 32118–32123.
- 42 M. Imai, K. Akiyama, T. Tanaka and E. Sano, *Compos. Sci. Technol.*, 2010, **70**, 1564–1570.
- 43 L. Jabbour, D. Chaussy, B. Eyraud and D. Beneventi, *Compos. Sci. Technol.*, 2012, **72**, 616–623.
- 44 B. Fugetsu, E. Sano, M. Sunada, Y. Sambongi, T. Shibuya, X. Wang and T. Hiraki, *Carbon*, 2008, **46**, 1256–1258.
- 45 M. Salajkova, L. Valentini, Q. Zhou and L. A. Berglund, *Compos. Sci. Technol.*, 2013, **87**, 103–110.
- 46 L. Hu, J. W. Choi, Y. Yang, S. Jeong, F. L. Mantia, L.-F. Cui and Y. Cui, *Proc. Natl. Acad. Sci. U.S.A.*, 2009, **106**, 21490–21494.
- 47 S. H. Yoon, H.-J. Jin, M.-C. Kook and Y. R. Pyun, *Biomacromolecules*, 2006, **7**, 1280–1284.
- 48 M. Agarwal, Q. Xing, B. S. Shim, N. Kotov, K. Varshney and Y. Lvov, *Nanotechnology*, 2009, **20**, 215602.
- 49 L. Hu, M. Pasta, F. L. Mantia, L. Cui, S. Jeong, H. D. Deshazer, J. W. Choi, S. M. Han and Y. Cui, *Nano Lett.*, 2010, **10**, 708–714.

