


Cite this: *RSC Adv.*, 2020, 10, 1287

# Hybrid polymers bearing oligo-L-lysine(carboxybenzyl)s: synthesis and investigations of secondary structure†

Merve Basak Canalp and Wolfgang H. Binder \*

Hybrid polymers of peptides resembling (partially) folded protein structures are promising materials in biomedicine, especially in view of folding-interactions between different segments. In this study polymers bearing repetitive peptidic folding elements, composed of N-terminus functionalized bis- $\omega$ -ene-functional oligo-L-lysine(carboxybenzyl(Z))s ( $\text{Lys}_n$ ) with repeating units ( $n$ ) of 3, 6, 12, 24 and 30 were successfully synthesized to study their secondary structure introduced by conformational interactions between their chains. The pre-polymers of ADMET, narrowly dispersed  $\text{Lys}_n$ s, were obtained by ring opening polymerization (ROP) of *N*-carboxyanhydride (NCA) initiated with 11-amino-undecene, following N-terminus functionalization with 10-undecenoyl chloride. The resulting  $\text{Lys}_n$ s were subsequently polymerized via ADMET polymerization by using Grubbs' first generation (G1) catalyst in 1,1,1,3,3,3-hexafluoroisopropanol (HFIP) generating the ADMET polymers (A- $[\text{Lys}_n]_m$ ) ( $m = 2$ –12) with molecular weights ranging from 3 to 28 kDa, displaying polydispersity ( $D$ ) values in the range of 1.5–3.2. After chemical analyses of  $\text{Lys}_n$ s and A- $[\text{Lys}_n]_m$ s by  $^1\text{H}$ -NMR, GPC and MALDI-ToF MS, secondary structural investigations were probed by CD spectroscopy and IR spectroscopy in 2,2,2-trifluoroethanol (TFE). In order to study A- $[\text{Lys}_n]_m$ s with defined molecular weights and low polydispersity values ( $D = 1.03$ –1.48), the ADMET polymers A- $[\text{Lys}_{n=3}]_{m=3}$  and A- $[\text{Lys}_{n=24}]_{m=4}$  were fractionated by preparative GPC, and subsequently analysed by  $^1\text{H}$ -NMR, analytical GPC, MALDI-ToF MS and CD spectroscopy. We can demonstrate the influence of chain length of the generated polymers on the formation of secondary structures by comparing  $\text{Lys}_n$ s with varying  $n$  values to the ADMET-polymers with the help of spectroscopic techniques such as CD and FTIR-spectroscopy in a helicogenic solvent.

Received 6th November 2019  
Accepted 21st December 2019

DOI: 10.1039/c9ra09189k

rsc.li/rsc-advances

## Introduction

Advances in the synthesis of biologically active macromolecules have paved the way for understanding the pathogenesis of many health issues such as cancer<sup>1</sup> and neurodegenerative diseases, *i.e.*, Alzheimer's<sup>2–5</sup> and Parkinson's,<sup>6</sup> in order to provide novel pharmaceuticals and drug/gene delivery systems for their treatments.<sup>7</sup> In particular, peptides and their conjugates that can meet the requirements of pharmacological activities because they form different reversible secondary structures, for instance  $\alpha$ -helix and  $\beta$ -sheet, induced by physicochemical changes, are of special relevance.<sup>8–12</sup> Thus, in the field of peptide chemistry, poly-L-lysine apparently holds importance for comprehending the native states of proteins, and thus can influence the drug discoveries for amyloidosis and siRNA/DNA deliveries.<sup>13,14</sup> While poly-L-lysine forms various secondary structures

depending on the physical and chemical environment, such as *e.g.*, pH,  $T$ , solvent, functional end/side groups and chain length,<sup>15–22</sup> hybrid polymers consisting of copolymers of this biologically compatible peptide can embody secondary structures with well-defined physical characteristics introducing additional stability, rigidity and flexibility into the overall polymeric system.<sup>23–26</sup>

Various hybrid polymers containing poly-L-lysine/lysine(Z) with defined architectures, *i.e.*, block copolymers, have been synthesized to integrate the biochemical features, *e.g.* pH responsivity,  $\alpha$ -helicity within the synthetic polymer chain, and therewith assess and control functionality of the assembly formation. Diblock copolymers of polyisoprene-block-poly-L-lysine/lysine(Z) have been synthesized by combination of anionic polymerization and ROP methods to obtain "hybrid" rod-coil block copolymers based on a synthetic coil segment and a polypeptide chain, displaying an  $\alpha$ -helical conformation for use in nanotechnology.<sup>27</sup> Poly(styrene)<sub>388</sub>-block-poly(L-lysine)<sub>138</sub> diblock copolymer changes its assembly to form pH responsive aggregate when mixed with a non-ionic surfactant, such as  $\text{C}_{12}\text{E}_6$ .<sup>28</sup> The secondary structure of poly(butadiene)<sub>107</sub>-block-poly(L-lysine)<sub>27</sub> has been probed by CD spectroscopy as

Faculty of Natural Science II (Chemistry, Physics and Mathematics), Martin Luther University Halle-Wittenberg, von-Danckelmann-Platz 4, Halle (Saale) D-06120, Germany. E-mail: wolfgang.binder@chemie.uni-halle.de

† Electronic supplementary information (ESI) available:  $^1\text{H}$ -NMR, MALDI-ToF, CD, IR, etc. See DOI: 10.1039/c9ra09189k



a function of pH and temperature, suggesting that an  $\alpha$ -helix of the poly(L-lysine) block transforms into  $\beta$ -sheets at higher temperature ( $T = 60\text{ }^{\circ}\text{C}$ ) at a pH = 11.2.<sup>29</sup> Another study has yet adjoined the secondary structural features of poly(L-lysine), combined with the thermo-responsive behaviour of poly(*N*-isopropylacrylamide) to achieve amphiphilic hybrid rod-coil block copolymers by combination of atom transfer radical polymerization (ATRP) and ROP.<sup>30</sup>

In this work it was aimed to combine the variable secondary structure of the oligo-L-lysine(Z) in combination with ethylene chains in a repetitive manner to investigate secondary structural changes such as  $\alpha$ -helicity induced assemblies of the so obtained hybrid polymers as represented in Fig. 1. Based on a combination of NCA ROP of *N*-carboxyanhydrides and ADMET polymerization techniques it is aimed to obtain hybrid polymers with the desired peptide/ethylene units, based on ADMET polymers of oligo-L-lysine(Z)s (green in Fig. 1). As the intermediate oligo-ethylene-segments (shown in grey, Fig. 1) can be considered as noninteracting in the sense of a supramolecular interaction, the so chosen molecular design allows to primarily study the intersegmental interactions of the (partially) folded oligo-L-lysine(Z)s, probed by CD- and IR-methods in TFE. In order to obtain reliable information as to the conformational status of the oligo-L-lysine(Z)s-segment, careful fractionation by preparative GPC was accomplished to enable a detailed study of the intersegmental conformational changes.

## Experimental section

### Materials and methods

DMF was dried over  $\text{CaH}_2$  and freshly distilled by applying vacuum (20 mbar) at  $50\text{ }^{\circ}\text{C}$ . Ethyl acetate was dried at  $100\text{ }^{\circ}\text{C}$  with  $\text{P}_2\text{O}_5$ . *N*-heptane was refluxed over sodium/benzophenone at  $120\text{ }^{\circ}\text{C}$ . All of the dried solvents were flushed with  $\text{N}_2$  gas before usage. All of the reagents were purchased from Sigma-Aldrich. All NMR-spectra were recorded on a Varian spectrometer (Gemini 200, Gemini 2000 and Unity 500) at 400 or 500 MHz at  $27\text{ }^{\circ}\text{C}$ . Trimethylsilane was used as internal standard. Deuterated chloroform ( $\text{CDCl}_3$ ) and dimethyl sulfoxide ( $\text{DMSO}-d_6$ ) were used as solvent. In the case of all polymer samples trifluoroacetic acid (TFA) (15% of volume) was added to the

deuterated chloroform ( $\text{CDCl}_3$ ). All chemical shifts ( $\delta$ ) were reported in parts per million (ppm) and the coupling constant ( $J$ ) in Hertz. For the interpretation of the spectra MestReNova v. 6.0.2 5475 was used. For analytical GPC measurements a Viscotek GPCmax VE 2001 with an HHR-H Guard-17369 and a GMHHR-N-18055 column in DMF at  $60\text{ }^{\circ}\text{C}$  was used. The sample concentration was  $5\text{ mg mL}^{-1}$ , the injection volume was  $100\text{ }\mu\text{L}$ . The detection was carried out *via* the refractive index with a VE 3580 RI detector of Viscotek at a temperature of  $35\text{ }^{\circ}\text{C}$  and a flow rate of  $1\text{ mL min}^{-1}$ . Polystyrene standards with molecular weights from 1 kDa to 115 kDa were used for external calibration. Preparative GPC was performed by a KD-2002.5 column from Shodex company attached on a VWR Hitachi Chromaster instrument using DMF (HPLC graded) as the eluent at  $55\text{ }^{\circ}\text{C}$  with a flow rate of  $0.70\text{ mL min}^{-1}$  injecting sample in a concentration of  $15\text{ mg mL}^{-1}$ . Refractive index detector from VWR at  $50\text{ }^{\circ}\text{C}$  was employed as the detector. The obtained data were analysed by using EZChrom Elite (version 3.3.2 SP2) software. ESI-ToF measurements were performed on a Focus micro ToF by Bruker Daltonics. The sample (1.00 mg) was dissolved in methanol (1.00 mL, HPLC grade) and directly infused ( $180.00\text{ }\mu\text{L h}^{-1}$ , positive or negative mode). MALDI-TOF MS measurements were carried out in either linear or reflector modes on a Bruker Autoflex III Smart beam equipped with a nitrogen laser source ( $\lambda = 337\text{ nm}$ ). The samples were dissolved in DMF (HPLC grade) ( $c = 5$  or  $10\text{ mg mL}^{-1}$ ) with the ration of 100 : 10 : 1 (matrix : analyte : salt). As matrix dithranol and/or DCTB with  $c = 10\text{ mg mL}^{-1}$  in THF, and as salt potassium trifluoroacetate (KTFA), sodium trifluoroacetate (NaTFA) and/or lithium trifluoroacetate (LiTFA) with  $c = 5\text{ mg mL}^{-1}$  in THF were used. For the data evaluations and simulation of the mass spectra of the polymer samples the computer programme Flex analysis (version 3.0) was used. CD spectroscopy measurements were performed with the instrument, JASCO Corp., J-810, Rev. 1.00, at a constant temperature ( $20\text{ }^{\circ}\text{C}$ ). The UV absorption was measured in CD units of millidegrees in the wavelength range of 260–190 nm. The cuvette cell used had a diameter of 0.1 cm. The correction of the measurements was done by subtraction of the absorption of the solvent, *i.e.*, 2,2,2-trifluoroethanol (TFE), from the absorbance of the sample which had a concentration of  $0.2\text{ mg mL}^{-1}$ . FTIR experiments were performed on a VERTEX 70 IR spectrometer from Bruker. For solid state investigations by FTIR spectroscopy a single reflex-diamond attenuated total reflectance unit was utilized. Samples prepared in TFE at different concentrations were investigated with a Specac Omni Cell demountable cell purchased from Sigma-Aldrich with  $\text{CaF}_2$  windows and a poly(tetrafluoroethylene) spacer which is 0.1 mm by measuring the pure solvent (TFE) as the background. The absorption bands were recorded in  $\text{cm}^{-1}$  in an area of 1500–1800  $\text{cm}^{-1}$ . For our data interpretation OPUS software was used.

### Synthesis of *N*-carboxyanhydride (NCA) monomer

The synthesis of NCA of *N*-carboxybenzyl(Z)-L-lysine has been reported in the literature and was performed similar to our previously reported work.<sup>22,31</sup>

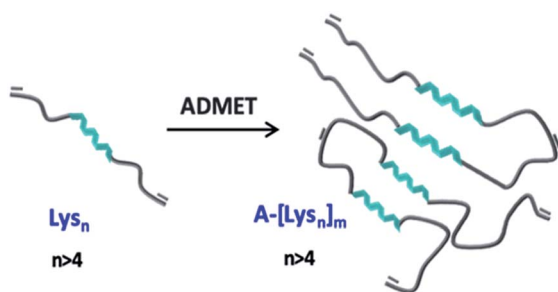


Fig. 1 Schematic representation of  $\alpha$ -helicity induced assemblies *via* intersegmental interactions of the oligo-L-lysine(Z)s-segments (green) along the ADMET-polymer, interrupted by oligo-ethylene-segments (grey).



Table 1 Synthetic data of ADMET polymerizations (GPC in DMF and polystyrene standards)

Entry	Pre-polymers (Lys <sub>n</sub> )	ADMET polymers (A-[Lys <sub>n</sub> ] <sub>m</sub> )	Solvent	Catalyst	DP <sub>NMR</sub> (n, m)	M <sub>n</sub> , GPC [kDa]	D
I	Lys <sub>n=3</sub>	A-[Lys <sub>n=3</sub> ] <sub>m=3</sub>	HFIP	G1	4, 3	3.7	2.2
II	Lys <sub>n=3</sub>	A2-[Lys <sub>n=3</sub> ] <sub>m=2</sub>	HFIP	G1	4, 2	2.9	1.5
III	Lys <sub>n=6</sub>	A-[Lys <sub>n=6</sub> ] <sub>m=12</sub>	HFIP	G1	6, 12	11.2	3.2
IV	Lys <sub>n=6</sub>	A2-[Lys <sub>n=6</sub> ] <sub>m=2</sub>	HFIP	G1	6, 2	3.7	1.4
V	Lys <sub>n=6</sub>	A3-[Lys <sub>n=6</sub> ] <sub>m=18</sub>	TFE	G1	6, 18	20.9	4.8
VI	Lys <sub>n=12</sub>	A-[Lys <sub>n=12</sub> ] <sub>m=8</sub>	HFIP	G1	12, 8	15.7	2.4
VII	Lys <sub>n=24</sub>	A-[Lys <sub>n=24</sub> ] <sub>m=4</sub>	HFIP	G1	24, 4	10.6	1.7
VIII	Lys <sub>n=24</sub>	A2-[Lys <sub>n=24</sub> ] <sub>m=3</sub>	HFIP	G1	24, 3	7.5	1.5
IX	Lys <sub>n=30</sub>	A-[Lys <sub>n=30</sub> ] <sub>m=7</sub>	HFIP	G1	30, 7	27.8	2.4

### Ring opening polymerization (ROP) of NCA monomer

ROP of NCA monomer was performed similar to the procedure with small modifications on the initiator to monomer ratio in order to achieve different degree of polymerizations.<sup>22,31</sup>

### N-terminus functionalization of mono-functional oligo-L-lysine(carboxybenzyl(Z))s (lys<sub>n</sub>)

The amidation reactions of N-terminus of oligo-L-lysine(Z)s were performed similar to the previously reported work (see ESI†).<sup>22</sup>

### ADMET polymerization of N-terminus functionalized oligo-L-lysine(carboxybenzyl(Z))s (A-[Lys<sub>n</sub>]<sub>m</sub>)

A general procedure was followed for the synthesis of ADMET polymers. A pre-dried Schlenk flask equipped with a stirrer filled with the respective pre-polymer Lys<sub>n</sub>. As a sample case, the pre-polymer Lys<sub>n=3</sub> (1000 mg) was dried under high vacuum for 72 hours before addition of small amount of HFIP (≈ 2 mL) under nitrogen atmosphere, just enough to dissolve Lys<sub>n=3</sub> at 45 °C. After 30 minutes, a trace amount of G1 (≈ 0.1 mg) was added to the reaction mixture under nitrogen atmosphere (N<sub>2</sub> atm) and let stir for 16 hours. Addition of trace amounts of G1 was repeated and after 2 hours of stirring under N<sub>2</sub> atm at 45 °C, gradual vacuum of 850 mbar to 700 mbar was applied continuously. Following this, the reaction medium was let stir under stable vacuum at 700 mbar for another 16 hours. Eventually the stirring was stopped due to high viscosity, therefore the viscous dark yellow/green substance was dissolved in DMF (≈ 1–2 mL) and subsequently precipitated in cold methanol (≈ 30 mL). The resulting dark yellow precipitate was washed several times with methanol and centrifuged until light yellow/white solid was obtained as the final product (A-[Lys<sub>n=3</sub>]<sub>m</sub>) (795 mg, yield: 70–80 wt%). <sup>1</sup>H-NMR analysis of polymer was prepared in a mixture of CDCl<sub>3</sub> and TFA (volume of 15%). Analytical GPC and MALDI-ToF MS samples were prepared in HPLC graded DMF (see Table 1, and ESI†).

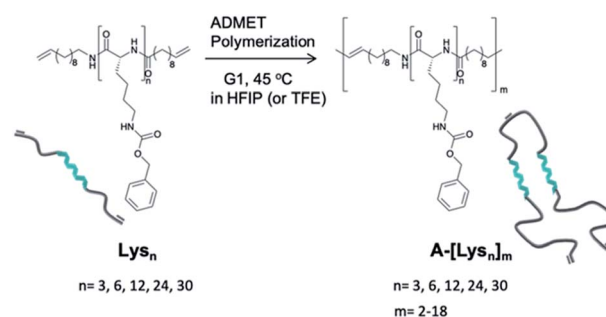
## Results and discussion

Our synthetic strategy is based on a combination of ring opening polymerization (ROP) of *N*-carboxyanhydride (NCA), followed by ADMET polymerization of appropriately bis-ene-functionalized oligomers to place the folding-elements of

Lys<sub>n</sub>s along a polymer chain, subsequently studying their intra- and interchain secondary structures. Acyclic diene metathesis (ADMET) polymerization<sup>32–34</sup> opens up possibilities to introduce different kinds of functional groups<sup>35–38</sup> into a polyethylene chain, introducing even anti-inflammatory drugs, such as ibuprofen and naproxen, into a polyethylene backbone.<sup>39</sup>

The ROP of NCA monomers is a viable and proven strategy to obtain artificial homo- and block co-peptides, where end group functionalized polymers can be obtained, given that the ROP mechanism follows the aspects of “livingness”.<sup>40,41</sup> Firstly, mono-functional oligo-L-lysine(Z)s with varying repeating units (*n* = 3, 6, 12, 24, 30) were synthesized *via* ROP of the lysine-NCA monomer using 11-aminoundecene as the primary initiator in DMF. It has been known that livingness and end group fidelity during ROP of NCAs can be achieved *via* the “Normal Amine Mechanism (NAM)” using primary or secondary amines as initiators.<sup>20,42–44</sup> Therefore the NCA monomer of *N*-carboxybenzyl(Z)-L-lysine amino acid, NCA(Z), was initiated *via* the primary amine, 11-aminoundecene, to obtain the mono-functional oligo-L-lysine(carboxybenzyl(Z))s in the first step. After proving the chain length and end groups *via* NMR- and MS-analyses (see ESI†), the N-terminal end groups were functionalized by direct amidation with 10-undecenoyl chloride *in situ* to achieve the respective bis-ω-ene-functional pre-polymers, Lys<sub>n</sub>s.<sup>22</sup>

Relying on the excellent possibilities of ADMET polymerization to include amino acids onto a polyolefin chain,<sup>45–47</sup> the Lys<sub>n</sub>s (*n* = 3, 6, 12, 24, 30) were further polymerized by ADMET polymerization in HFIP by addition of Grubb's first generation



Scheme 1 Schematic representation of the synthetic pathway of ADMET polymerization of Lys<sub>n</sub><sup>22</sup> by using Grubb's first catalyst (G1).



Table 2 Summary of all obtained data of oligo-L-lysine(Z)s ( $\text{Lys}_n$ ), their ADMET polymers ( $\text{A}[\text{Lys}_n]_m$ )s and the selected fractions

Entry	Samples	$\text{DP}_{\text{theo.}} (n)$	$\text{DP}_{\text{NMR}} (n, m)$	$M_n$ , MALDI [kDA]	$M_n$ , GPC in DMF [kDA]	$\bar{D}$	CD-Spec. (in TFE)
<b>Oligo-L-lysine(Z)s</b>							
I	$\text{Lys}_{n=3}$	3	4	$2.2 \pm 1.8$	1.9	1.06	$\beta$ II turn
II	$\text{Lys}_{n=6}$	6	6	$3.2 \pm 1.6$	2.8	1.17	25% $\alpha$ -helicity
III	$\text{Lys}_{n=12}$	12	11	$3.3 \pm 1.7$	3.3	1.36	27% $\alpha$ -helicity
IV	$\text{Lys}_{n=24}$	24	23	$3.6 \pm 1.2$	6.3	1.40	53% $\alpha$ -helicity
V	$\text{Lys}_{n=30}$	30	30	$5.6 \pm 4$	8.0	1.33	65% $\alpha$ -helicity
<b>ADMET polymers of oligo-L-lysine(Z)s &amp; the selected fractions</b>							
VI	$\text{A}[\text{Lys}_{n=3}]_m=3$	3	3	$2.5 \pm 1.2$	3.7	2.20	$\beta$ II turn
VII	F10 + 11 + 12	3	NA	NA	23.2	1.21	18% $\alpha$ -helicity
VIII	F13 + 14	3	NA	$8.1 \pm 3$	12.8	1.12	15% $\alpha$ -helicity
IX	F15	3	NA	$6.5 \pm 2$	9.2	1.08	$\beta$ II turn
X	F17	3	NA	$2.8 \pm 1.7$	5.8	1.05	$\beta$ II turn
XI	F18	3	NA	$2.6 \pm 1.6$	4.7	1.05	$\beta$ II turn
XII	F21 + 22 + 23	3	NA	$2.6 \pm 0.5$	2.6	1.04	$\beta$ II turn
XIII	F24 + 25	3	NA	$2.1 \pm 0.5$	2.1	1.03	$\beta$ II turn
XIV	$\text{A}[\text{Lys}_{n=6}]_m=12$	6	12	$4.3 \pm 2.5$	11.2	3.21	32% $\alpha$ -helicity
XV	$\text{A}[\text{Lys}_{n=12}]_m=8$	12	8	$4.7 \pm 3.2$	15.7	2.43	44% $\alpha$ -helicity
XVI	$\text{A}[\text{Lys}_{n=24}]_m=4$	24	4	$4.1 \pm 2.4$	10.6	1.74	69% $\alpha$ -helicity
XVII	F12	24	NA	NA	23.2	1.16	77% $\alpha$ -helicity
XVIII	F13	24	4	NA	19.1	1.15	62% $\alpha$ -helicity
XIX	F15	24	NA	$14.2 \pm 2$	12.1	1.07	53% $\alpha$ -helicity
XX	F16	24	NA	$13.7 \pm 2$	10.9	1.07	52% $\alpha$ -helicity
XXI	F18	24	NA	$7.7 \pm 2$	7.1	1.08	40% $\alpha$ -helicity
XXII	$\text{A}[\text{Lys}_{n=30}]_m=7$	30	7	$4.3 \pm 2$	27.8	2.41	79% $\alpha$ -helicity

catalyst (G1) (see Scheme 1 and Table 1). All obtained  $\text{Lys}_n$ s and  $\text{A}[\text{Lys}_n]_m$ s were characterized by  $^1\text{H}$ -NMR, MALDI-ToF MS and analytical GPC (see Table 2 and ESI†). Additionally, the ADMET polymers  $\text{A}[\text{Lys}_{n=3}]_m=3$  and  $\text{A}[\text{Lys}_{n=24}]_m=4$  were fractionated by

preparative GPC and then selectively investigated by MALDI-ToF MS, analytical GPC,  $^1\text{H}$ -NMR (see Table 2, Fig. 2, 3 and ESI†). Structural investigations of  $\text{Lys}_n$ s,  $\text{A}[\text{Lys}_n]_m$ s and the selected fractions of  $\text{A}[\text{Lys}_n]_m$ s were mainly done by CD

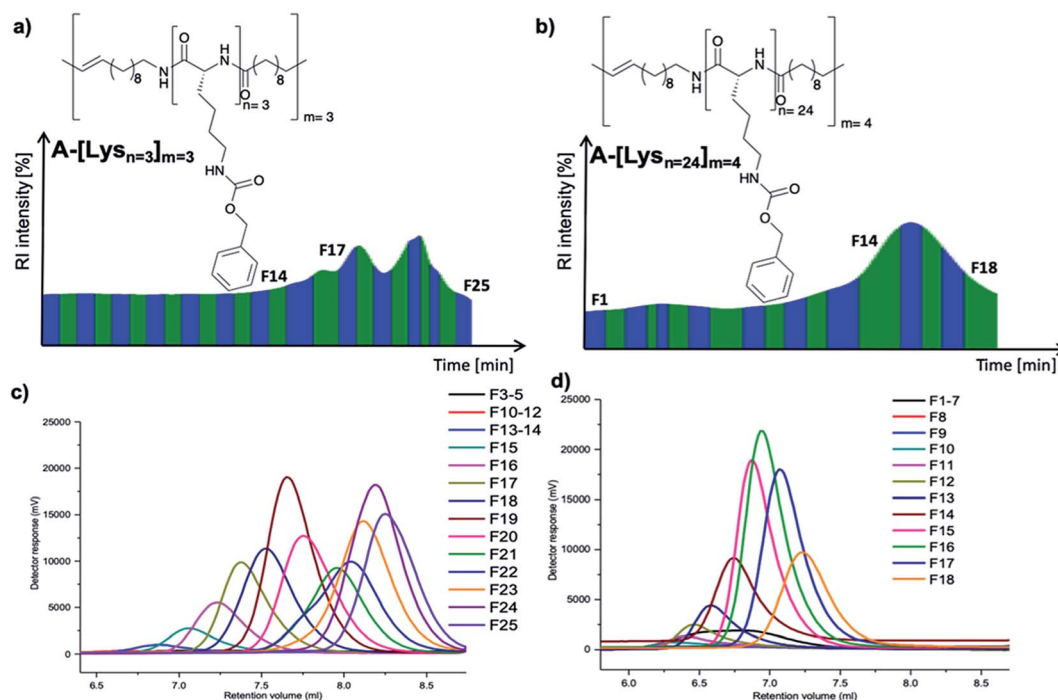


Fig. 2 Preparative GPC graphs of (a)  $\text{A}[\text{Lys}_{n=3}]_m=3$  (b)  $\text{A}[\text{Lys}_{n=24}]_m=4$  along with their analytical GPC results of respective (c) fractions of  $\text{A}[\text{Lys}_{n=3}]_m=3$  and (d) fractions of  $\text{A}[\text{Lys}_{n=24}]_m=4$ .





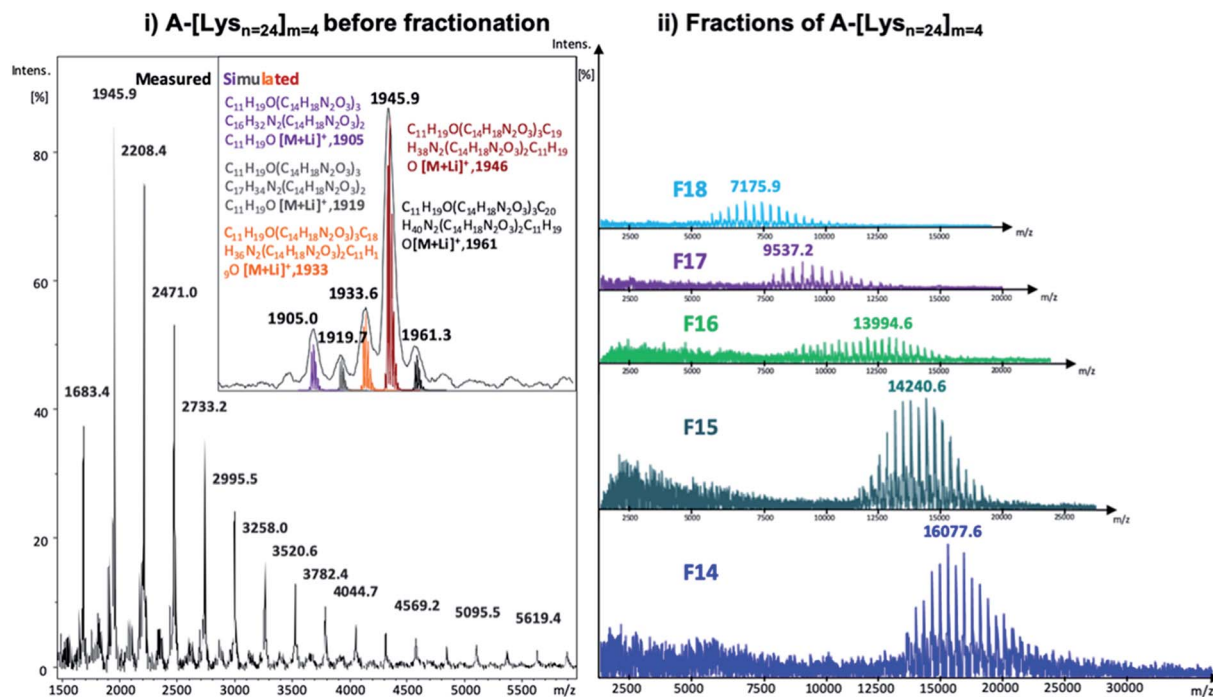


Fig. 3 MALDI-ToF MS spectra of (i) A-[Lys<sub>n=24</sub>]<sub>m=4</sub> before fractionation along with the simulated isotopic patterns (reflector mode) and (ii) the selected fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (linear mode) (also see ESI†).

spectroscopy in TFE (see Table 2, Fig. 4, 5 and also ESI†). IR spectroscopy in TFE was also probed to analyse secondary structure features of Lys<sub>n</sub>s and A-[Lys<sub>n</sub>]<sub>m</sub>s (see Fig. 6 and ESI†).

### Fractionation of ADMET polymers by preparative GPC

With ADMET being a polycondensation reaction, the obtained polymers display polydispersity (PDI) values in the range of 1.5–3.2. As the chain length dependent folding studies requires smaller PDIs, the fractional analyses of selected ADMET polymers were accomplished with the help of preparative GPC of

these hybrid polymers, proving their low polydispersity (*D*) values (see Table 2, and also ESI Tables 1S and 2S†).

Thus, ADMET polymers A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (Table 2, entry VI) and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Table 2, entry XVI) were separated into fractions (Table 2, entries VII–XIII for the former and Table 2, entries XVII–XXI for the latter) according to their molecular weights, thus displaying significantly lower *D* values than their native unfractionated samples *via* preparative GPC technique in DMF at a column temperature of 55 °C.

Preparative GPC traces of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (Fig. 2a) and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Fig. 2b) along with the analytical GPC results of their fractions (Fig. 2c and d respectively) are demonstrated. As depicted in Fig. 2a and b, all fractions are named as F<sub>x</sub>, *i.e.*, *x* is the alignment number of the fraction trace in the preparative GPC graph. The complete analytical GPC results of all the

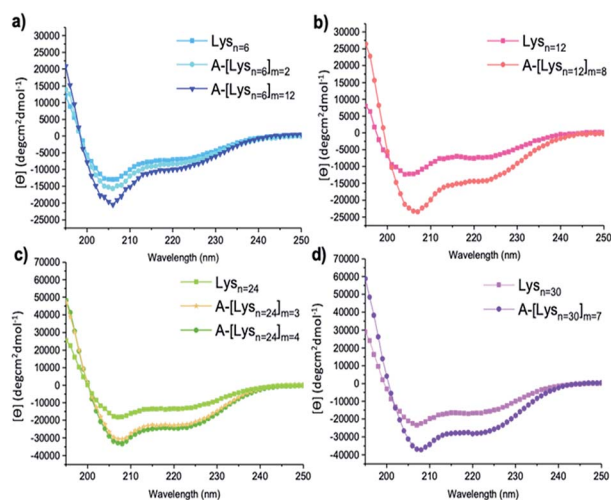


Fig. 4 CD spectra of (a) Lys<sub>n=6</sub>, A-[Lys<sub>n=6</sub>]<sub>m=2</sub>, A-[Lys<sub>n=6</sub>]<sub>m=12</sub> (b) Lys<sub>n=12</sub>, A-[Lys<sub>n=12</sub>]<sub>m=8</sub> (c) Lys<sub>n=24</sub>, A-[Lys<sub>n=24</sub>]<sub>m=3</sub>, A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (d) Lys<sub>n=30</sub> and A-[Lys<sub>n=30</sub>]<sub>m=7</sub>. (c = 2 mg mL<sup>-1</sup> in TFE).

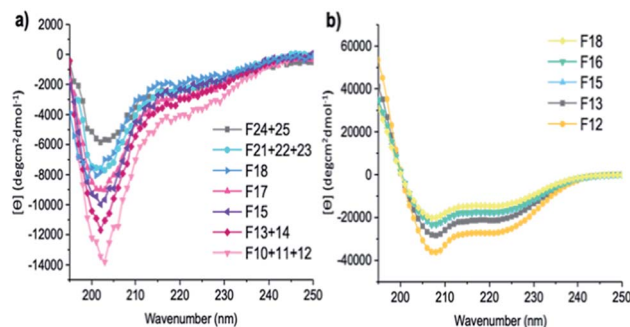


Fig. 5 CD spectra of fractions of (a) A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (F10 + 11 + 12, F13 + 14, F15, F17, F18, F21 + 22 + 23, F24 + 25) and (b) A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (F12, F13, F15, F16, F18). (c = 2 mg mL<sup>-1</sup> in TFE).



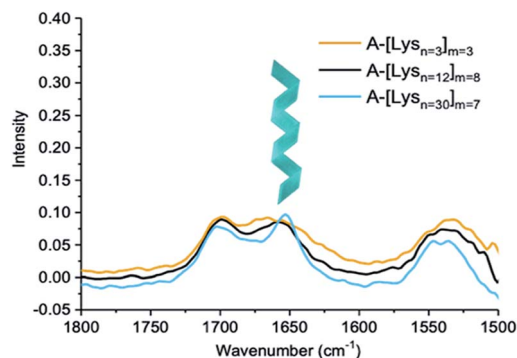


Fig. 6 IR spectra of ADMET polymers A-[Lys<sub>n=3</sub>]<sub>m=3</sub>, A-[Lys<sub>n=12</sub>]<sub>m=8</sub> and A-[Lys<sub>n=30</sub>]<sub>m=7</sub> (*c* = 5 mg mL<sup>-1</sup> in TFE).

fractions of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (Table 1S†) and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Table 2S†) are present within the ESI.†

As can be seen from the obtained analysis results, molecular weights of fractions of ADMET polymer A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (Table 2, entry VI) increases with the decreasing number of the fraction trace number, *e.g.* F25 (*M<sub>n</sub>* ≈ 2 kDa) and F15 (*M<sub>n</sub>* ≈ 10 kDa) (see ESI, Table 1S†). All obtained fractions of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> measured by analytical GPC in DMF display low *D* values (1.20 > *D* > 1.03) which proves the successful separation of the fractions.

In the case of the high molecular weight ADMET polymer, A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Table 2, entry XVI), molecular weight of the fractions also increases with the decreasing number of the fraction trace number, *e.g.* F18 (Table 2, entry XXI) (*M<sub>n</sub>* ≈ 7.2 kDa) and F10 (*M<sub>n</sub>* ≈ 23 kDa) (see ESI, Table 2S†). All obtained fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> measured by analytical GPC in DMF have low *D* values (1.48 > *D* > 1.07) when they are compared with ADMET polymers which have *D* values over 2.0 in general.

In Fig. 3, the measured MALDI-ToF MS spectra of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> before fractionation with the simulated isotopic patterns (Fig. 3i) and after fractionation of selected fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub>, *i.e.*, F14–F18 (Fig. 3ii) are shown. The measured MALDI spectra and the respective simulated isotopic patterns of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Fig. 3i) prove the ADMET polymer structure, having -CH<sub>2</sub> characteristic difference due to the fragmented isomers of ADMET polymers, where the simulated and the measured spectra match. Similar to the analytical GPC investigations, the measured MALDI spectra of selected fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> in Fig. 3ii verify the increasing molecular weight of the fractions with decreasing fraction trace number obtained by preparative GPC (see Fig. 2 and also ESI, Table 2S†). According to the measured MALDI spectra in Fig. 3ii, F14 has a molecular weight of ≈ 16 ± 3 kDa whereas the low molecular weight fraction F18 has a molecular weight of ≈ 7 ± 3 kDa, which evidences an average molecular weight difference of 9 kDa. Therefore, all these fractional analyses of ADMET polymers are suited for the subsequent secondary structural investigations.

### Conformational investigations by spectroscopic methods in TFE

In this work, the analyses of the secondary structures of the oligomers and their ADMET derivative polymers were

conducted in an  $\alpha$ -helix promoting solvent 2,2,2-trifluoroethanol (TFE) which stabilizes  $\alpha$ -helix structures due to its high polarity.<sup>48,49</sup> We aimed to investigate the effects of number of repeating units (*n*) present in oligomers and in the ADMET polymers on the formation of secondary structures, *i.e.*, their  $\alpha$ -helix propensities, as measured by CD spectroscopy in TFE. The summary of all CD spectroscopy measurements along with their analytical GPC results of oligomers, ADMET polymers and their selected fractions are presented within Table 2. The percentages of  $\alpha$ -helicities are presented in Table 2 and were calculated according to literature.<sup>50</sup> (see ESI, eqn (1S)†) The secondary structural investigations of Lys<sub>n</sub>s along with their ADMET polymer derivatives A-[Lys<sub>n</sub>]<sub>m</sub>s are shown in Fig. 4. Selected fractions of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> are depicted in Fig. 5.

Enhanced  $\alpha$ -helicity with increasing *n* number was observed for oligo-L-lysine(Z)s Lys<sub>n</sub>s (see Table 2, and ESI Fig. 20S†) as also supported by our previous investigations.<sup>22</sup> As an example, Lys<sub>n=6</sub> (Table 2, entry II) displays 25%  $\alpha$ -helicity, while Lys<sub>n=30</sub> (Table 2, entry V) with its higher degree of polymerization displays 65%  $\alpha$ -helicity. Similarly, structural investigations utilizing CD spectroscopy have also been probed by Huesmann *et al.*<sup>20</sup> where mono-functional poly-L-lysine(Z)s prepared by hexylamine with *n* = 24, 57, 87, 196 (*D<sub>s</sub>* ~ 1.70) have been investigated in HFIP. According to their investigations, poly-L-lysine(Z) with *n* = 5–15 did not display an ordered secondary conformation, however only at a chain length of *n* ≈ 60  $\alpha$ -helicity was observed, thus showing an increase in helicity with increasing *n*.

Secondly, the effects resulting from repetition of the helical segments on the secondary structure formation was investigated by the comparison of the oligomers and their respective ADMET polymer derivatives. Thus, the measured CD spectra of the oligomers along with their respective ADMET polymers are given in Fig. 4. In this regard, since the helical segments are repetitively present within the hybrid polymers of A-[Lys<sub>n</sub>]<sub>m</sub>s (*n* > 4), it is observed from the increasing intensities of [ $\theta$ ]<sub>222</sub> from Fig. 4 that their  $\alpha$ -helicity values increase from around 30 to 80 percentage (see Table 2, entries XIV, XV, XVI and XXII). To exemplify, Lys<sub>n=24</sub> (Table 2, entry IV) displays ~ 53%  $\alpha$ -helicity, and on the other hand the respective ADMET polymer of Lys<sub>n=24</sub>, A-[Lys<sub>n=24</sub>]<sub>m=4</sub> (Table 2, entry XVI), displays around 69%  $\alpha$ -helicity. ADMET polymer of Lys<sub>n=30</sub>, A-[Lys<sub>n=30</sub>]<sub>m=7</sub> (Table 2, entry XXII), in a similar way, shows a higher tendency to form helical structure which increases from 65 to 79%  $\alpha$ -helicity. Both results point at an interference of the individual Lys<sub>n</sub> – blocks along the ADMET-polymers, leading to a distinct increase of their helicity.

It also was of interest to investigate the effects of the chain lengths of the ADMET polymers on the formation of  $\alpha$ -helices. Therefore, selected fractions of ADMET polymers, namely A-[Lys<sub>n=3</sub>]<sub>m=3</sub> (Table 2, entry VI) and A-[Lys<sub>n=24</sub>]<sub>m=4</sub>, were examined by CD spectroscopy in TFE. From the CD measurements represented in Fig. 5 and the calculations of  $\alpha$ -helicity percentage values given in Table 2, one can deduce that the overall molecular weight of the hybrid polymers, and thus, the repeating unit of L-lysine(Z) moieties plays a decisive role in



forming  $\alpha$ -helices. Before fractional analyses, as it was expected, ADMET polymer A-[Lys<sub>n=3</sub>]<sub>m=3</sub>, did not show any  $\alpha$ -helicity due to its low number of L-lysine(Z) units, devoid of a stabilization of a proper helical turn.<sup>51,52</sup> However, after fractionation of A-[Lys<sub>n=3</sub>]<sub>m=3</sub>,  $\alpha$ -helicity could be observed from CD spectroscopy measurements done in TFE, where the obtained fractions F10–12 (Table 2, entry VII) and F13–14 (Table 2, entry VIII) show around 15–18%  $\alpha$ -helicity as it is also shown in Fig. 5a. Furthermore, some selected fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> were also investigated by CD as shown in Fig. 5b. The signal intensities at 222 nm,  $[\theta]_{222}$ , get even more significant for the fractions of A-[Lys<sub>n=24</sub>]<sub>m=4</sub> as the molecular weight of the fraction increases from F18 (Table 2, entry XXI) to F12 (Table 2, entry XVII), again proving that  $\alpha$ -helicity propensities increase with the increasing molecular weight of the hybrid polymer.

Last but not least, the molecular weight dependence on the secondary structural features of the selected fractions of both A-[Lys<sub>n=3</sub>]<sub>m=3</sub> and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> could also be compared. For instance, F10 + 11 + 12 (Table 2, entry VII) and F12 (Table 2, entry XVII) display the same  $M_n$  values (23.2 kDa), however the former one shows only around 18%  $\alpha$ -helicity whereas the latter has around 77%  $\alpha$ -helicity. Similarly, comparing the  $\alpha$ -helicity percentage values of F13 + 14 (Table 2, entry VIII) with F15 (Table 2, entry XIX) and F15 (Table 2, entry IX) with F16 (Table 2, entry XX) indicates that not only the molecular weights of the hybrid polymers but also the number of repeating units of L-lysine(Z) along the whole chain plays a significant role in their secondary structural properties, *i.e.*, their  $\alpha$ -helicity propensities.

Furthermore, infrared (IR) spectroscopy, which is a well-known and versatile experimental technique to analyse chemical as well as biological samples either in solution or in the solid state, was additionally employed.<sup>19,53,54</sup> With the help of IR spectroscopy measurements, it was intended to gain insight into the presence of the other common secondary structures, *e.g.*,  $\beta$ -sheets and  $\beta$ -turns. However, owing to the sufficient solubility of only some of the samples, we could obtain structural investigations by IR spectroscopy in only partial depth (see ESI†).

With the help of IR, one can observe secondary structures which produce characteristic electron densities in the amide C=O groups, giving rise to distinctive amide I frequencies (1620–1700 cm<sup>-1</sup>) in the IR spectra. Since the length of the hydrogen bond resulting from  $\alpha$ -helix conformation formed by the C=O group of the residue  $n$  positioned to interact intramolecularly with the N-H group of the residue  $n + 4$  will be slightly longer (and thus weaker) than that of in a  $\beta$ -sheet structure, it gives further increase in amide I frequency.<sup>53,55</sup> Concordantly,  $\alpha$ -helix structures contribute to amide I frequency signals at around 1648–1660 cm<sup>-1</sup>, while  $\beta$ -sheet structures are observed at around 1625–1640 cm<sup>-1</sup>.<sup>56</sup>

IR spectra of ADMET polymers, A-[Lys<sub>n=3</sub>]<sub>m=3</sub> are shown with a yellow line, A-[Lys<sub>n=12</sub>]<sub>m=8</sub> with a black line and A-[Lys<sub>n=30</sub>]<sub>m=7</sub> with a blue line are shown in Fig. 6. As previously observed from CD investigations in TFE,  $\alpha$ -helicity structure formation tendency increases with the increasing number of repeating units ( $n$ ) which could be also observed from the intense sharp

peak of the blue line at  $\approx 1650$  cm<sup>-1</sup> proving  $\alpha$ -helicity of the ADMET polymer (A-[Lys<sub>n=30</sub>]<sub>m=7</sub>) in IR spectra. Since the number of repeating units of peptide decreases, as seen in the cases of ADMET polymers, A-[Lys<sub>n=12</sub>]<sub>m=8</sub> and A-[Lys<sub>n=3</sub>]<sub>m=3</sub>, the IR signal at  $\approx 1650$  cm<sup>-1</sup> get weaker and broader most probably because of the presence of other unordered structure. The signals at around 1700 cm<sup>-1</sup> stem from the dynamic nature of secondary structure formation resulting in different secondary structural features such as  $\beta$ -sheets and  $\beta$ -turns.

In order to work up a connection in between secondary structures of different types of hybrid polymers such as linear peptides with  $\beta$ -turn motifs in TFE<sup>57</sup> and amyloid-type fibrils with  $\beta$ -sheet structures in solution, CD and IR methods have been used jointly.<sup>58</sup> Consequently, IR measurements assist the secondary structural analyses performed by CD spectroscopy, while the latter provides more accurate estimations of  $\alpha$ -helix content and the former one is more sensitive to  $\beta$ -sheets.<sup>59</sup> In comparison to IR measurements, our CD investigations performed in TFE help us quantify  $\alpha$ -helicity of the oligomers and the ADMET polymers straightforwardly as shown in Fig. 4 and 5b and Table 2. Hence, in an attempt to indicate the presence of  $\beta$ -sheets and  $\beta$ -turns, especially in the case of highly  $\alpha$ -helical polymers, *e.g.*, A-[Lys<sub>n=30</sub>]<sub>m=7</sub>, we employed IR as shown in Fig. 6. Similarly, another study has utilized both IR and CD methods and reported that at native pH, lipopeptides form  $\beta$ -sheet nanofibers in aqueous solution with a strong IR peak at 1616–1620 cm<sup>-1</sup> and CD spectra displaying similar characteristic signals of a  $\beta$ -sheet secondary structure, *i.e.*, minima at  $\sim 216$  nm and a maximum at  $\sim 197$  nm.<sup>60</sup>

To this date, the features determining the folding behaviours of helix bundles, *e.g.*, four-helix bundle motifs, have been drawn upon as a model for understanding the forces governing functional protein folding such as in the case of transmembrane protein GLUT1.<sup>61–63</sup> According to the findings the burial of hydrophobic residues, conformational entropy, packing constraints, helix dipole and interhelical turns have an impact on forming such helix–helix bundles. Additionally, it has been known that the perfectly ordered biologically active three-dimensional structures of proteins form such functional assemblies by the interplay of varying interactions, *e.g.*, non-covalent interactions.<sup>64,65</sup> These noncovalent cooperative interactions are mainly interpreted from their hydrogen bonding affinities.<sup>66,67</sup> Similar to our current study model helical polymers, studied under different conditions to examine their helix–coil transitions, also showed changes in secondary structure induced by the repeating number of helical chains, *i.e.*, length and strength of helices.<sup>68–70</sup> Based on CD and IR-measurements in our study we can also prove a change of helical structures induced by chain length ( $n$ ), end chain moieties and helicogenic solvent effects. In summary, our secondary structural investigations in TFE revealed that ADMET polymers have higher propensities than the oligomers to exhibit and stabilize  $\alpha$ -helical structure, proposed as a result of the packing of their repetitive helical segments present within the hybrid polymer chain that form intramolecular interactions of the hydrogen bonding moieties, namely L-lysine(Z)s, once the  $\alpha$ -helix structure has been provided ( $n > 4$ ).





## Conclusions

We here aimed to generate bio-mimicking macromolecules with repetitive helical segments that could serve as models to investigate their structural properties in a reductionist approach manner. Therefore, we successfully synthesized and characterized hybrid polymers, ADMET polymers (A-[Lys<sub>n</sub>]<sub>m</sub>) of bis- $\omega$ -ene-functional oligo-L-lysine(Z)s (Lys<sub>n</sub>), with different repeating units ( $n = 3, 6, 12, 24, 30$ ). We separated ADMET polymers, A-[Lys<sub>n=3</sub>]<sub>m=3</sub> and A-[Lys<sub>n=24</sub>]<sub>m=4</sub>, into their fractions by using preparative GPC technique in DMF to access lower polydispersity values with controlled molecular weights. All obtained oligo-L-lysine(Z)s, ADMET polymers, and selectively chosen fractions of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> were analysed by <sup>1</sup>H-NMR, analytical GPC and MALDI-ToF MS.

The secondary structural investigations were conducted with the help of CD spectroscopy and IR spectroscopy in TFE. Throughout our secondary structural investigations, we intended to draw attention to the effects of number of chains ( $n$ ) of Lys<sub>n</sub>s, as reported in our preceding work, and of their ADMET derivatives, *i.e.*, A-[Lys<sub>n</sub>]<sub>m</sub>s, in an  $\alpha$ -helix promoting solvent. Our secondary structural investigations have shown that an increasing number of peptide unit increases the propensity to form  $\alpha$ -helices. This phenomenon can be better understood by comparing the CD spectra of the fractions of A-[Lys<sub>n=3</sub>]<sub>m=3</sub> and A-[Lys<sub>n=24</sub>]<sub>m=4</sub> selectively investigated by CD spectroscopy in TFE, proving the effects of the controlled molecular weights with low  $\bar{D}$  values of the hybrid polymers on the formation of  $\alpha$ -helices. Thus, the individual oligomers, the Lys<sub>n</sub> segments, do interact along the main chain, inducing increased helicity with increasing chain length of the overall ADMET polymer. It can be proposed that this type of cooperative phenomenon is similar to other multi-folded proteins, where close proximity between individual segments can change the overall conformation of individual part – *e.g.* in globular proteins with helix-helix interaction motifs, as common for many transmembrane proteins. We think that these polymers can serve as a model-system to study influences from distance-related folding and refolding phenomena.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

We thank the grant Project A03 nr. 189853844-TRR 102 (Polymers under multiple constraints) for financial support from the German Research Foundation (DFG) *via* the research centre SFB/TRR 102 (Project A3 and A12). We also thank a grant from the “Leistungszentrum Chemie und Biosystemtechnik des Landes Sachsen-Anhalt” for financial support, internal project S1 (Binder). We would like to thank Prof. Hauke Lilie from the Institute of Biochemistry and Biotechnology, Martin Luther University Halle-Wittenberg for allowing to work with the CD spectroscopy instrument.

## Notes and references

- 1 Y. K. Sung and S. W. Kim, *Cancer Med.*, 2019, **2**, 6–13.
- 2 R. N. Kalara and S. I. Harik, *J. Neurochem.*, 1989, **53**, 1083–1088.
- 3 M. M. M. P. W. Hruz, *Mol. Membr. Biol.*, 2001, **18**, 183–193.
- 4 M. Mueckler and C. Makepeace, *Biochemistry*, 2009, **48**, 5934–5942.
- 5 D. Deng, C. Xu, P. Sun, J. Wu, C. Yan, M. Hu and N. Yan, *Nature*, 2014, **510**, 121.
- 6 C. Giordano, D. Albani, A. Gloria, M. Tunesi, S. Batelli, T. Russo, G. Forloni, L. Ambrosio and A. Cigada, *Int. J. Artif. Organs*, 2009, **32**, 836–850.
- 7 N. Alasvand, S. Kargozar, P. B. Milan, N. P. S. Chauhan and M. Mozafari, in *Adv. Func. Polym. for Biomed. Apps.*, ed. M. Mozafari and N. P. Singh Chauhan, Elsevier, 2019, pp. 275–299.
- 8 M. Fändrich and C. M. Dobson, *EMBO J.*, 2002, **21**, 5682–5690.
- 9 A. Lavasanifar, J. Samuel and G. S. Kwon, *Adv. Drug Delivery Rev.*, 2002, **54**, 169–190.
- 10 A. Lalatsa, A. G. Schätzlein, M. Mazza, T. B. H. Le and I. F. Uchegbu, *J. Controlled Release*, 2012, **161**, 523–536.
- 11 A. Makino, *Polym. J.*, 2014, **46**, 783.
- 12 N. Liu, B. Li, C. Gong, Y. Liu, Y. Wang and G. Wu, *Colloids Surf., B*, 2015, **136**, 562–569.
- 13 J. S. Choi, E. J. Lee, Y. H. Choi, Y. J. Jeong and J. S. Park, *Bioconjugate Chem.*, 1999, **10**, 62–65.
- 14 P. Liu, H. Yu, Y. Sun, M. Zhu and Y. Duan, *Biomaterials*, 2012, **33**, 4403–4412.
- 15 A. I. Arunkumar, T. K. S. Kumar, T. Sivaraman and C. Yu, *Int. J. Biol. Macromol.*, 1997, **21**, 299–305.
- 16 A. I. Arunkumar, T. K. S. Kumar and C. Yu, *Int. J. Biol. Macromol.*, 1997, **21**, 223–230.
- 17 O. Kambara, A. Tamura, A. Naito and K. Tominaga, *Phys. Chem. Chem. Phys.*, 2008, **10**, 5042–5044.
- 18 Ł. Szyc, S. Pilarz and B. Czarnik-Matusiewicz, *J. Mol. Liq.*, 2008, **141**, 155–159.
- 19 A. Mirtič and J. Grdadolnik, *Biophys. Chem.*, 2013, **175**–**176**, 47–53.
- 20 D. Huesmann, A. Birke, K. Klinker, S. Türk, H. J. Räder and M. Barz, *Macromolecules*, 2014, **47**, 928–936.
- 21 K. Cieřlik-Boczula, *Biochimie*, 2017, **137**, 106–114.
- 22 M. B. Canalp, A. Meister and W. H. Binder, *RSC Adv.*, 2019, **9**, 21707–21714.
- 23 A. Carlsen and S. Lecommandoux, *Curr. Opin. Colloid Interface Sci.*, 2009, **14**, 329–339.
- 24 J.-P. Chen, I. M. Chu, M.-Y. Shiao, B. R.-S. Hsu and S.-H. Fu, *J. Ferment. Bioeng.*, 1998, **86**, 185–190.
- 25 M. Metzke, N. O'Connor, S. Maiti, E. Nelson and Z. Guan, *Angew. Chem.*, 2005, **44**, 6529–6533.
- 26 K. T. Al-Jamal, W. T. Al-Jamal, J. T. W. Wang, N. Rubio, J. Buddle, D. Gathercole, M. Zloh and K. Kostarelos, *ACS Nano*, 2013, **7**, 1905–1917.
- 27 J. Babin, J. Rodriguez-Hernandez, S. Lecommandoux, H.-A. Klok and M.-F. Achard, *Faraday Discuss.*, 2005, **128**, 179–192.





- 28 G. Orts Gil, M. Łosik, H. Schlaad, M. Drechsler and T. Hellweg, *Langmuir*, 2008, **24**, 12823–12828.
- 29 K. E. Gebhardt, S. Ahn, G. Venkatachalam and D. A. Savin, *J. Colloid Interface Sci.*, 2008, **317**, 70–76.
- 30 C.-J. Huang and F.-C. Chang, *Macromolecules*, 2008, **41**, 7041–7052.
- 31 G. J. M. Habraken, M. Peeters, C. H. J. T. Dietz, C. E. Koning and A. Heise, *Polym. Chem.*, 2010, **1**, 514–524.
- 32 K. B. Wagener, J. M. Boncella and J. G. Nel, *Macromolecules*, 1991, **24**, 2649–2657.
- 33 H. Li, L. Caire da Silva, M. D. Schulz, G. Rojas and K. B. Wagener, *Polym. Int.*, 2016, DOI: 10.1002/pi.5188.
- 34 J. A. Smith, K. R. Brzezinska, D. J. Valenti and K. B. Wagener, *Macromolecules*, 2000, **33**, 3781–3794.
- 35 F. Marsico, M. Wagner, K. Landfester and F. R. Wurm, *Macromolecules*, 2012, **45**, 8511–8518.
- 36 T. Lebarbé, A. S. More, P. S. Sane, E. Grau, C. Alfes and H. Cramail, *Macromol. Rapid Commun.*, 2014, **35**, 479–483.
- 37 S. Reimann, U. Baumeister and W. H. Binder, *Macromol. Chem. Phys.*, 2014, **215**, 1963–1972.
- 38 J. T. Offenloch, J. Willenbacher, P. Tzvetkova, C. Heiler, H. Mutlu and C. Barner-Kowollik, *Chem. Commun.*, 2017, **53**, 775–778.
- 39 J. K. Leonard, D. Turek, K. B. Sloan and K. B. Wagener, *Macromol. Chem. Phys.*, 2010, **211**, 154–165.
- 40 H. R. Kricheldorf, *Angew. Chem., Int. Ed.*, 2006, **45**, 5752–5784.
- 41 Y. Shen, X. Fu, W. Fu and Z. Li, *Chem. Soc. Rev.*, 2015, **44**, 612–622.
- 42 H. H. James, in *Ring-Opening Polymerization*, American Chemical Society, 1985, ch. 5, vol. 286, pp. 67–85.
- 43 W. Zhao, Y. Gnanou and N. Hadjichristidis, *Biomacromolecules*, 2015, **16**, 1352–1357.
- 44 W. Vayaboury, O. Giani, H. Cottet, A. Deratani and F. Schué, *Macromol. Rapid Commun.*, 2004, **25**, 1221–1224.
- 45 T. E. Hopkins, J. H. Pawlow, D. L. Koren, K. S. Deters, S. M. Solivan, J. A. Davis, F. J. Gómez and K. B. Wagener, *Macromolecules*, 2001, **34**, 7920–7922.
- 46 T. E. Hopkins and K. B. Wagener, *Macromolecules*, 2003, **36**, 2206–2214.
- 47 J. K. Leonard, T. E. Hopkins, K. Chaffin and K. B. Wagener, *Macromol. Chem. Phys.*, 2008, **209**, 1485–1494.
- 48 P. Pengo, L. Pasquato, S. Moro, A. Brigo, F. Fogolari, Q. B. Broxterman, B. Kaptein and P. Scrimin, *Angew. Chem.*, 2003, **115**, 3510–3514.
- 49 R. Rajan and P. Balaram, *Int. J. Pept. Protein Res.*, 1996, **48**, 328–336.
- 50 K.-S. Krannig, J. Sun and H. Schlaad, *Biomacromolecules*, 2014, **15**, 978–984.
- 51 K. Olsen and J. Bohr, *Theor. Chem. Acc.*, 2010, **125**, 207–215.
- 52 B. Haimov and S. Srebnik, *Sci. Rep.*, 2016, **6**, 38341.
- 53 A. Barth, *Biochim. Biophys. Acta, Bioenerg.*, 2007, **1767**, 1073–1101.
- 54 M. Rozenberg and G. Shoham, *Biophys. Chem.*, 2007, **125**, 166–171.
- 55 W. K. Surewicz, H. H. Mantsch and D. Chapman, *Biochemistry*, 1993, **32**, 389–394.
- 56 M. Jackson and H. H. Mantsch, *Crit. Rev. Biochem. Mol. Biol.*, 1995, **30**, 95–120.
- 57 M. Hollósi, Z. Majer, A. Z. Rónai, A. Magyar, K. Medzihradsky, S. Holly, A. Perczel and G. D. Fasman, *Biopolymers*, 1994, **34**, 177–185.
- 58 M. J. Krysmann, V. Castelletto and I. W. Hamley, *Soft Matter*, 2007, **3**, 1401–1406.
- 59 K. A. Oberg, J. M. Ruyschaert and E. Goormaghtigh, *Eur. J. Biochem.*, 2004, **271**, 2937–2948.
- 60 V. Castelletto, A. Kaur, R. M. Kowalczyk, I. W. Hamley, M. Reza and J. Ruokolainen, *Biomacromolecules*, 2017, **18**, 2013–2023.
- 61 J. M. Pascual, D. Wang, R. Yang, L. Shi, H. Yang and D. C. De Vivo, *J. Biol. Chem.*, 2008, **283**, 16732–16742.
- 62 S. Kamtekar and M. H. Hecht, *FASEB J.*, 1995, **9**, 1013–1022.
- 63 A. Kauko, L. E. Hedin, E. Thebaud, S. Cristobal, A. Elofsson and G. von Heijne, *J. Mol. Biol.*, 2010, **397**, 190–201.
- 64 M. J. Williams and M. Bachmann, *Polymers*, 2016, **8**, 245.
- 65 A. S. Mahadevi and G. N. Sastry, *Chem. Rev.*, 2016, **116**, 2775–2825.
- 66 Y. Zhou, G. Deng, Y.-Z. Zheng, J. Xu, H. Ashraf and Z.-W. Yu, *Sci. Rep.*, 2016, **6**, 36932.
- 67 Y. Li, Y. Wang, G. Huang and J. Gao, *Chem. Rev.*, 2018, **118**, 5359–5391.
- 68 Y. Ren, H. Fu, R. Baumgartner, Y. Zhang, J. Cheng and Y. Lin, *ACS Macro Lett.*, 2017, **6**, 733–737.
- 69 Y. Ren, R. Baumgartner, H. Fu, P. van der Schoot, J. Cheng and Y. Lin, *Biomacromolecules*, 2017, **18**, 2324–2332.
- 70 Y.-D. Wu and Y.-L. Zhao, *J. Am. Chem. Soc.*, 2001, **123**, 5313–5319.

